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Energy-efficient resistive switching synaptic devices based on patterned Ag nanotriangles with tunable gaps fabricated using plasma-assisted nanosphere lithography†

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The development of synaptic devices featuring metallic nanostructures with brain-analog hierarchical architecture, capable of mimicking cognitive functionalities, has emerged as a focal point in neuromorphic computing. However, existing challenges, such as inconsistent and unpredictable switching, high voltage requirements, unguided filament formation, and detailed fabrication processes, have impeded technological progress in the domain. The present study addresses some of these challenges by leveraging periodic nanostructures of Ag fabricated *via* plasma-assisted nanosphere lithography (NSL). The triangular nanostructures with a preferred orientation offer enhanced localized electric fields, facilitating low voltage electromigration at the sharp edges to guide predictive filament formation. A thorough investigation into gap control between the nanostructures through oxygen plasma treatment enables the attainment of an optimized low switching voltage of 0.86 V and retention at an ultra-low current compliance of 100 nA. The optimized device consumes low power, typically in the fJ range, akin to biological neurons. Furthermore, the device showcases intriguing synaptic characteristics, including controlled transition from short- to long-term potentiation, associative learning, *etc.*, projecting its potential in perceptual learning, memory formation, and brain-inspired computing. COMSOL Multiphysics simulation, supported by *ex situ* electron microscopic imaging, confirms the controlled and predictable filament formation facilitated by electric field enhancement across the strategic nanostructures. Thus, the work highlights the potential of NSL-based cost-effective fabrication techniques for realizing efficient and biomimetic synaptic devices for neuromorphic computing applications.

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Introduction

In recent years, there has been tremendous progress in fabricating microelectronic devices to emulate brain functionalities, including short-term plasticity (STP), long-term potentiation (LTP), associative learning, and other cognitive processes.^{1–6} Such behavioral functionalities are born out of the ability of the active synaptic junction *via* accumulation and firing of the action potential. In the domain of artificial devices, such functionality is derived from the conductance state (and such properties) achieved by the device based on the strength and the sequence of arrival of inputs, much the way a memristor functions.^{7–9} Such a unique property has been realized in

many ways through resistive switching and spintronics devices.^{10–12}

Among diverse resistive switching synaptic devices, namely electrochemical metallization (ECM), valence change memory (VCM), phase change memory (PCM), *etc.*,^{13–17} ECM stands out for its exceptional control over filament formation and ultra-low power consumption.^{18,19} ECM is an electric field-driven phenomenon related to the atomic connection between nanostructures, where the gap between the nanostructure or nanocluster is crucial for resistive switching.^{20,21} There are various reports on gap-dependent electromigration-based devices.^{22–24} For instance, electron beam lithography (EBL)-fabricated Au nanostructure with a 54 to 62 nm gap was reported by Sakai and coworkers for the ECM-based conduction to control tunnelling resistance across the nanogaps.²⁵ Bose *et al.* have worked on controllable switching of a self-assembled nanoparticle network, leading to atomic connections between the interparticle gaps.²⁶ In another instance, for effective localization of electrons at the tip, Pt nano-junctions of gaps less

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than 2 nm have been fabricated using nanoimprint lithography, but the gaps were found to suffer with the migration of atoms under applied voltage.²⁷ These systems used high voltages (>2 V) for the electromigration of metallic atoms between nanogap and complex lithographic fabrication. Concurrently, previous studies from our group have explored nanoscale Ag agglomerates formed through the dewetting of an Ag film. In such devices, electromigration occurs within the gaps of these Ag agglomerates at low voltages, enabling basic synaptic functions like STP, LTP, and associative learning with lower energy requirements. Furthermore, ECM-based devices demonstrate excellent device-to-device consistency, making them highly suitable for integration into crossbar array architectures.^{28,29} These self-formed structures offer stochastic filamentary-based cognitive functionalities.^{30–34}

Achieving nanoscale structure with tunable nanogaps is crucial for ECM-based devices. However, fabricating such structures for synaptic devices typically demands laborious techniques like photolithography, EBL, *etc.*,³⁵ which are intricate, costly, and time-intensive.³⁶ In contrast, nanosphere lithography (NSL) offers a simple, scalable and inexpensive alternative for generating large-area array of 2D nanostructures with precise control over shape, size, and interparticle spacing.^{37–39} This process relies on a disordered colloidal system that adopts the lowest available entropy to develop a self-assembled, ordered, and stable structure.³⁸ Noble metal nanostructures produced *via* NSL have found applications in diverse fields, such as optical chemo-sensors and biosensors platforms.^{40–42} However, there are hardly any reports on its utilization for neuromorphic device fabrication.

In this work, we have fabricated two-terminal planar synaptic devices with Ag triangular nanostructures using NSL. Scanning electron microscopy (SEM) and transmission electron microscopy (TEM) have been employed to extract microscopic information pertaining to fabricated nanostructures. The metal bowtie gap was adjusted *via* an oxygen plasma treatment (OPT), allowing control over the nanosphere shadow to yield adjustable gaps between the nanostructures, thereby facilitating lowering of the threshold switching while also reducing power requirement per synapse. Furthermore, the device emulates controlled STP to LTP transition and association learning aided by an ECM-based filament formation, which is further corroborated *via* simulating the electric field distribution across nanostructures using COMSOL Multiphysics and an *ex situ* SEM. COMSOL simulation confirms that the electromigration of Ag follows the path between the bowtie tip where the electric field is localized. The study is the first of its kind using the scalable NSL method for neuromorphic device fabrication.

Experimental section

Device fabrication using nanosphere lithography

Initially, glass substrates of dimension $3.75 \times 1.25 \text{ cm}^2$ were ultrasonically cleaned successively in soap solution, deionized

(DI) water, acetone, and isopropyl alcohol for 10 min each and then dried using N_2 gas. A 10% polystyrene (PS) sphere solution (Bang Laboratory Inc.) with a 500 nm diameter was used for the NSL after dilution with ethanol at 1 : 1 v/v%. At first, a Petri dish of diameter 5 cm was filled with water, and then the cleaned glass slide was placed inside at an angle of 45° . Next, a 10 μL pipette was used for adding the PS solution dropwise at the air–water interface. A bluish-green layer, confirming PS monolayer formation, appeared on the surface of the water after dropping 40 μL of the solution. Thereafter, another glass slide was used for transferring the monolayer on the glass slide. Subsequently, OPT (Henniker Plasma, HPT-100) was performed on the template to reduce the size of the PS sphere to the desired dimension. The OPT was performed for different durations from 60 to 300 s (30 s intervals) at a power of 100 W while keeping the oxygen flow rate at 20 sccm. Afterward, the monolayer was used as a shadow mask for Ag deposition. A thermal evaporator system (HHV ATS-500) was used to deposit 60 nm Ag. The deposition was carried out at a rate of 1.5 nm s^{-1} with the substrate holder rotation at 8 rev. per min to obtain a quasi-triangular (QT) structure. No rotation was used to form intact Ag triangles. Consequently, the PS monolayer was removed by using scotch tape. To complete the device fabrication, 100 nm thick Al electrodes of dimension $5 \times 2 \text{ mm}^2$ were deposited on the Ag nanostructure by using a shadow mask of 40 μm wide Cu wire.

Characterization and measurements

TEM (Thermo Fisher, TALOS F200S G2, 200 keV) was performed to observe the Ag nanostructure and its planer orientation. For imaging purpose, the Ag triangles from the glass were first lifted off using IPA drops and then transferred onto a TEM grid. Field emission scanning electron microscope (FESEM) images were taken using Apreo-2 SEM (Thermo Fisher Scientific Inc.). UV-Vis absorption spectra were collected using a spectrometer from Agilent (Cary Series). A source measure unit (SMU) (Keithley 2461) connected to two tungsten probes attached to an XYZ stage controller (Thorlabs Inc.) was used for all the electrical measurements, namely *I*–*V* sweeps, and pulsing. Dual *I*–*V* sweeps were taken with a proper time gap between the successive sweeps to avoid non-volatile switching of the device. ImageJ software was used to calculate the fill factor and nanogaps from the FESEM image analysis. COMSOL Multiphysics 6.1 was used to simulate the electric field distribution by using the AC/DC module. First, the geometry was fed using a binary image of the device and converted into a curve, and then the material corresponding to that was assigned. Then, the simulation was performed with extremely fine mesh.

Results and discussion

Fabrication of the periodic nanostructure of Ag triangles involves fewer steps, as demonstrated schematically in Fig. 1a (see Experimental section for details). At first, PS nanoparticles

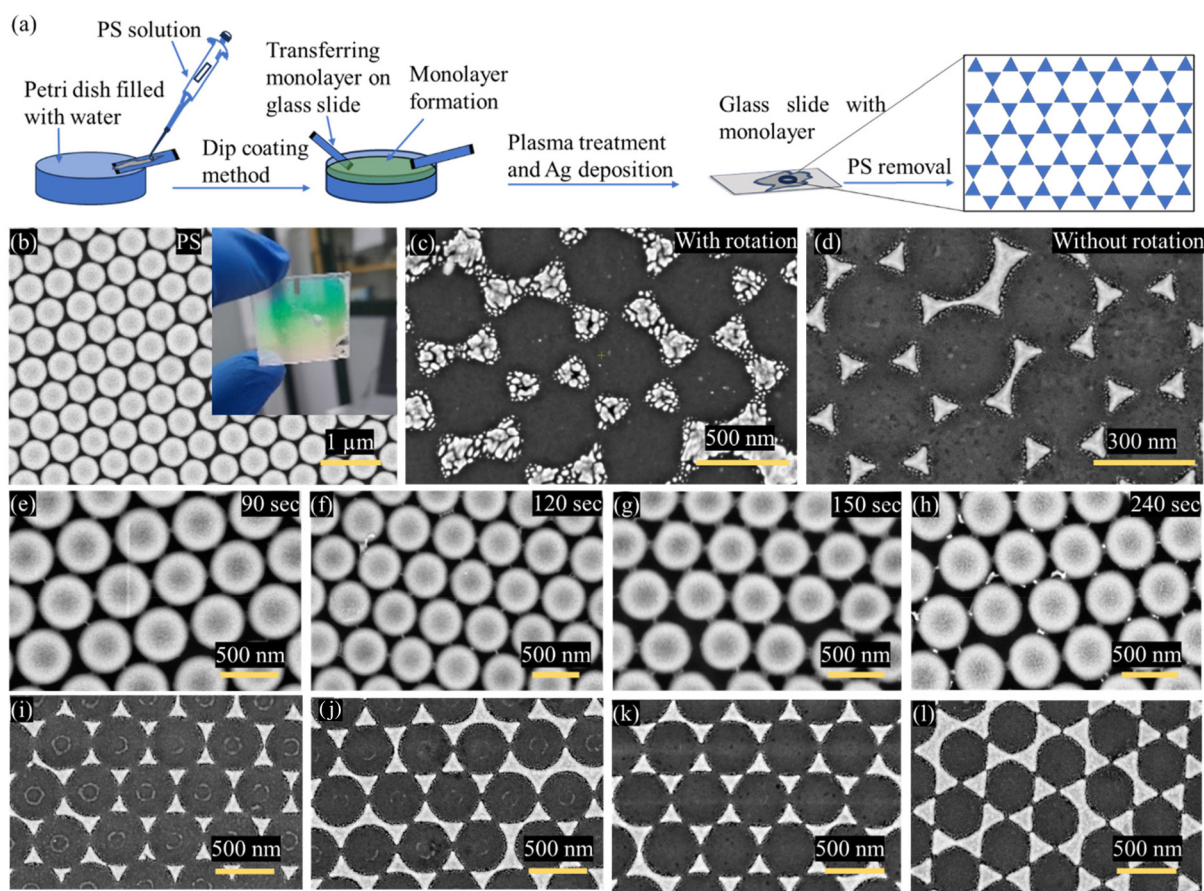


Fig. 1 (a) Schematic illustrating the fabrication process of Ag triangular nanostructure by NSL. The process includes the Marangoni formation of a PS monolayer onto the water surface, followed by its transfer onto a glass substrate, partial plasma etching of PS, Ag deposition, and PS removal. FESEM images of (b) the PS monolayer template; inset showing the photograph of the template at a large scale. Ag nanostructures obtained (c) with and (d) without rotation of substrate holder during Ag deposition; (e), (f), (g), and (h) ordered structure of PS monolayers after OPT for 90, 120, 150, and 240 s, respectively; (i), (j), (k) and (l) respective Ag nanostructures after PS removal.

of diameter $\sim 472.2 \pm 10.2$ nm (see the distribution profile in Fig. S1a†) were dispersed in 1 : 1 (v/v%) ethanol–water solution and then slowly spread onto the surface of water in a Petri dish. Nanoparticles self-assembled into a monolayer (see the FESEM image in Fig. 1b) due to the surface tension gradient (Marangoni effect) between water and the PS solution,⁴³ causing the Marangoni force to move the suspended particles in an outward direction until covering the water surface completely. Next, the layer was dip-transferred onto a pre-cleaned glass substrate, as shown in the inset of Fig. 1b, portraying a uniform film with diffracted colors emanating from its periodic structure. Afterward, the monolayer, either as formed or after OPT, was used as a shadow mask for Ag deposition. Thereafter, the PS monolayer was removed to attain the Ag triangular nanostructure. The deposition was carried out in two ways – (i) with and (ii) without substrate holder rotation during thermal evaporation. While the former produced quasi-triangular (QT) nanostructures consisting of tiny discrete particles due to a glancing angle deposition (Fig. 1c), the latter resulted in intact triangular structures (see Fig. 1d) with varied gaps depending upon the OPT time (T_p).⁴⁴ Without the OPT,

triangles were largely spaced (as seen in Fig. 1d) with a gap ascribed to a larger shadow masking at the inter-nanosphere interfaces.⁴⁵ The OPT leads to a reduction in the shadowing effect (Fig. 1e–h), for $T_p = 90, 120, 150,$ and 240 s, respectively due to the PS size reduction⁴⁵ (see Fig. S1b–e†), resulting in a decrement in nanogap between triangles, as depicted in the FESEM images in Fig. 1i–l. The PS spheres reduce their size (see Fig. S1f†) by reacting with the oxygen ions while liberating gases such as CO, CO₂, etc.⁴⁶ The rod-like nanobridge between the PS spheres has appeared with the OPT due to the melting and etching of the PS spheres with T_p .⁴⁷ An increase in T_p shrinks the PS spheres and thins and breaks the nanobridge contact, leading to the non-closed-packed structure, as shown in Fig. 1e–h. Bright-contrast TEM images were captured to better visualize the Ag nanostructure and investigate the presence of any unwanted impurities (Fig. S2a–c†). The images along with the lattice parameters calculated from the selected area electron diffraction (SAED) pattern confirm the presence of metallic Ag (Fig. S2d and Table S1†) without impurities.

To better understand the NSL in obtaining control over the structural parameter, the variation in Ag fill factor (shown as 1

– FF) and the corresponding average nanogap ' d ' were plotted against T_p values. ImageJ software was utilized to calculate the FF (as exemplified in Fig. 2a and Fig. S3a†) and ' d ' (see Fig. S3b and Table S2†). The plot in Fig. 2b indicates that the OPT leads to a linear reduction in both ' d ' and ' $1 - FF$ ' of Ag at rates of 0.4 nm s^{-1} and $0.06\% \text{ s}^{-1}$, respectively. As T_p increases, the size and shadowing effect of PS spheres decreases, and thus, the FF increases. UV-VIS absorbance spectroscopy was also performed to gain insight into the plasmonic effect of the periodic nanostructure. As depicted in Fig. 2c and d, the structure with $d \sim 78 \text{ nm}$ shows a localized surface plasmon (LSPR) peak at 660 nm , which undergoes a red shift with the decreasing d . The red shift of the plasmonic peak originates from the coupling of plasma electrons between the Ag triangles.⁴⁸ As the nanogap becomes smaller, the coupling between adjacent triangles increases, resulting in an easy exchange of electrons within the bowtie structure, thereby lowering the effective restoring force. As the force decreases, the resonant energy to induce the plasmonic coupling reduces, causing a red shift in the LSPR dipolar peak.⁴⁹

The unique bowtie structures were exclusively exploited in this study in fabricating neuromorphic devices, as schematically demonstrated in Fig. 3a, where the device analogy to the biological synapse is showcased. The release of neurotransmitters across the synapse between the pre and post-neurons helps in memory formation by adjusting synaptic weight in the biological systems.⁵⁰ As depicted, the metallic triangular

islands and the intermittent gaps in the device act as neurons and synapses, respectively. Also, the device conductance update serves as the synaptic weight in the device. The device containing a network of such synapse-like junctions resembles, in principle, the biological neural network structurally. The device fabrication was completed by forming Al electrodes spaced $40 \mu\text{m}$ apart on the periodic nanostructures (Fig. 3a). The unperturbed device exhibits an initial high resistance state (HRS) attributed to the presence of unconnected nanogaps between the electrodes. During an $I-V$ sweep, the device transitions to low resistance state (LRS), with a high ON/OFF ratio (10^4), while device current being regulated by a specific current compliance (I_{CC}) (Fig. 3b–e). Such transition could be attributed to the electric field aided electromigration of Ag across nanogaps.⁵¹ Due to the unipolar nature of the device, it has a nearly symmetric $I-V$ sweep for positive and negative polarities, as shown in Fig. S4.† With the gradual withdrawal of voltage, the device automatically returns to the HRS state without requiring an opposite bias. When an opposite bias is applied, the device switches back to the LRS state. But LRS can be erased to the HRS by applying the voltage within a tiny window ($\sim 200 \text{ mV}$) around the origin (red box in Fig. S4.†). Further analysis related to the filament formation and its correlation to the structural geometry of Ag is detailed later.

To begin with, the $I-V$ sweeps were performed at a low I_{CC} (100 nA) for each device (Fig. S5a.†), resulting in higher

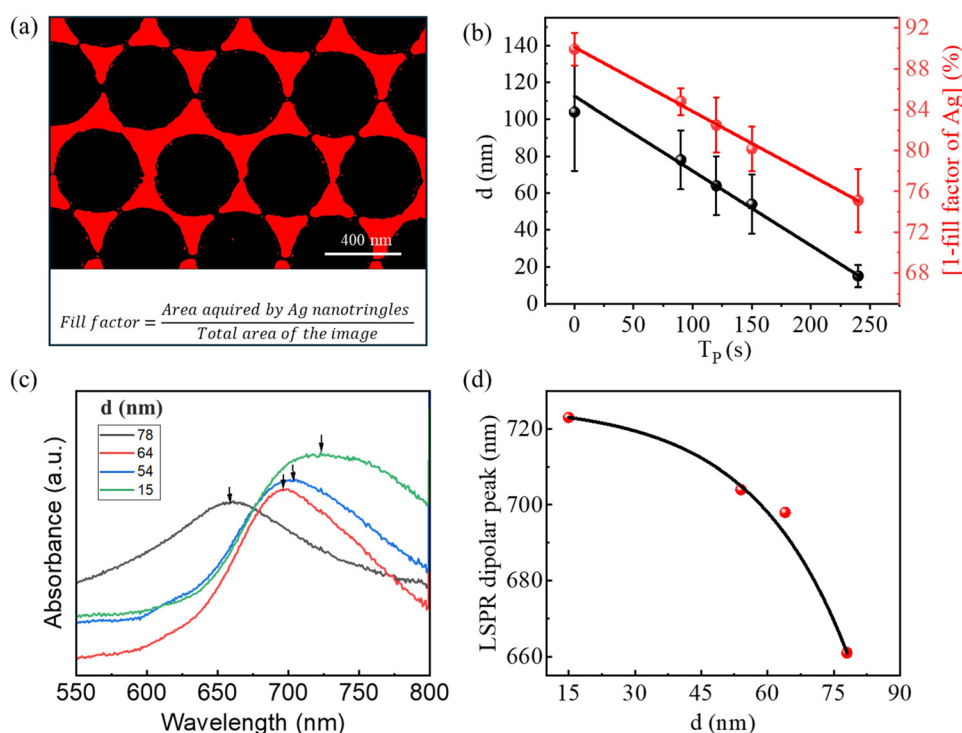


Fig. 2 (a) Ag fill factor (FF) calculation from FESEM image. (b) Average nanogap ' d ' and FF variation (shown as $1 - FF$) with OPT time (T_p), (c) absorption spectra of Ag superlattice structures prepared by following T_p of 90, 120, 150, and 240 s. (d) Variation of the corresponding LSPR dipolar peak with the nanogap.

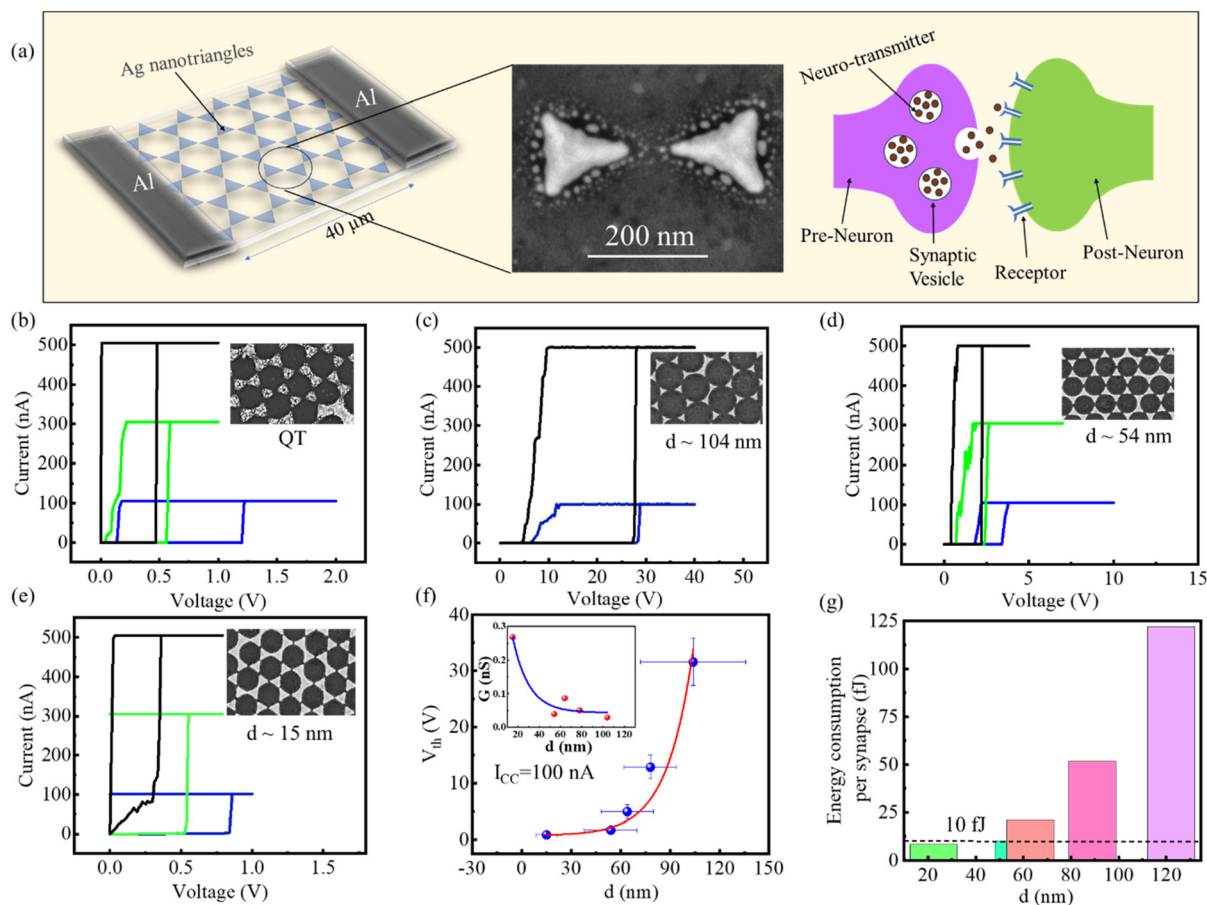


Fig. 3 (a) Schematic of the device with a magnified bowtie structure resembling biological synapses. I - V characteristics of samples having (b) QT structure, (c) $d \sim 104$ nm, (d) $d \sim 54$ nm, and (e) $d \sim 15$ nm, depicting a decreasing tendency of V_{th} with increasing I_{CC} . (f) V_{th} variation with ' d '. Inset shows the HRS conductance exponential decay with increase in nanogap at $I_{CC} = 100$ nA. (g) Histogram comparing energy consumption per synapse between devices with different nanogaps.

forming voltages during HRS to LRS switching (see Fig. S5b† and histogram for the sequence of a sweep at each I_{CC} in Fig. S5c†). The pristine devices initially require a high voltage to form a filamentary path. However, with repeated sweeps, the switching or threshold voltage decreased significantly (see Fig. S5e-i†). For a comprehensive analysis, I - V sweeps of several devices comprising QT and bowtie structures with different nanogaps are compared, as shown in Fig. 3b-e. Interestingly, the device with the lowest gap of 15 nm shows an ultra-low switching voltage (V_{th}) of only 300 mV at an I_{CC} of 500 nA. The lowering of V_{th} could be attributed to the unique structural design of the nano-architecture, which helps in enhancing and concentrating the electric field during the sweep (later verified by COMSOL simulation). A similar V_{th} value was also obtained for the QT device. Nonetheless, the QT structure (Fig. S6a†) suffered from geometrical instability (see Fig. S6b†) upon repeated sweeping (Fig. S6c†), probably due to its scattered nature, leading to a drastic fluctuation in V_{th} , as shown in Fig. S6d.† Repeated voltage sweeps for all devices with different nanogaps are shown in Fig. S5d-i.† It is intriguing that V_{th} shows a decreasing trend with

increasing I_{CC} , which may be due to a better filament consolidation assisted by enhanced Joule heating at higher I_{CC} .³¹ Device-to-device performance was examined using four devices ($d \sim 15$ nm) named D1, D2, D3, and D4, as shown in the insets of Fig. S7b-e.† Initially, I - V sweeps were performed at an I_{CC} of 100 nA, revealing higher forming voltages (Fig. S7a†). The switching voltage of devices drops with successive sweeps (1, 2, and 3) at each I_{CC} value, 100, 300, and 500 nA (shown in Fig. S7b-e†). Apart from higher forming voltage, the devices exhibited lesser spread in threshold voltages, as depicted in Fig. S7f.† Furthermore, as seen in Fig. 3f, an exponentially increasing correlation is observed between the average V_{th} and ' d '. Also, the V_{th} varies exponentially with nanogap number density (n/A), which was calculated using ImageJ software (see Fig. S8†). For a deeper analysis, the conductance of these devices in the HRS states was calculated from the slope of the I - V sweep till the threshold switching (see the inset in Fig. 3f). The result indicates an exponential decrement in the HRS conductance with d , inferring the device is in a tunneling regime; otherwise, a linear behavior is expected.⁵² The direct

tunneling current ' I ' decreases exponentially with the gap ' d ', as given by the following equation,⁵²

$$I = A \times V \times \exp(-B \times d) \quad (1)$$

where A and B are constants, and V is the applied bias; for a fixed I_{CC} , I is a constant, and hence the above equation can be written as,

$$V = a \times \exp(B \times d) \quad (2)$$

where a is another constant.

Thus, V_{th} increases exponentially with the gap in the low I_{CC} range (~ 100 nA), where the conduction transpires majorly through tunneling. The stability study demonstrates that the device exhibits consistent threshold-switching behavior even after one year of storage under ambient conditions (see Fig. S9†).

A detailed analysis was performed to obtain per-synapse energy requirement, as shown in Fig. 3g, indicating an exponential increment in the energy consumption with the nanogap (see Fig. S10b†). The calculation (as shown in Note S1†) considers two probable synaptic connections (indicated by arrows 2 and 3 in Fig. S10a†) for each triangle. Remarkably, the device with $d \sim 15$ nm consumes only 8.4 fJ per synapse, close to the limit of a biological synapse (10 fJ) and outperforming many literature reports.³¹

To emulate brain functionality such as weightage-dependent STP to LTP transition, appropriate voltage pulses were applied to devices after obtaining V_{th} from the I - V sweeps. The weightage was varied by using I_{CC} of different amplitudes.

Pulses with amplitudes of 2, 3, and 6 V (with 10 pulses of pulse width, $t_w = 50$ ms and pulse interval, $t_i = 50$ ms) were applied to devices with a QT structure, $d \sim 68$ and 54 nm, respectively (Fig. 4a-c, respectively). The conductance retention behavior of the device after pulsing was monitored with a reading voltage (V_R) of 10 mV. Intriguingly, with the increasing

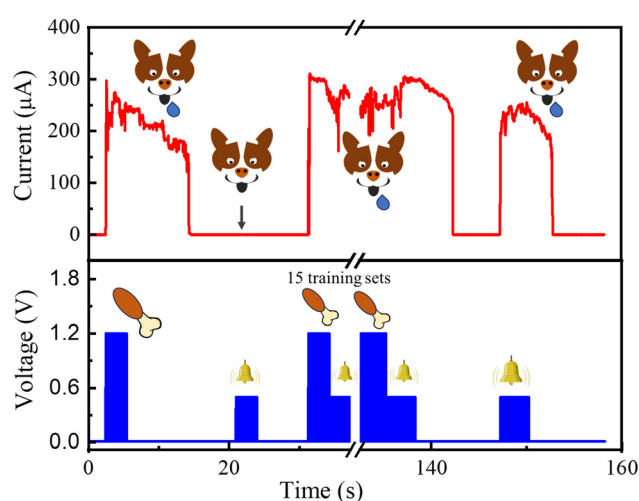


Fig. 5 Emulation of classical conditioning using the device subjected to an appropriate training set of food (1.2 V) and bell (0.5 V) pulses. Before training, the response was observed only with the food pulse, whereas it was absent for the bell pulse. After training with 15 consecutive sets of food and bell pulses, the device responded to the bell pulse alone thereby mimicking associative learning.

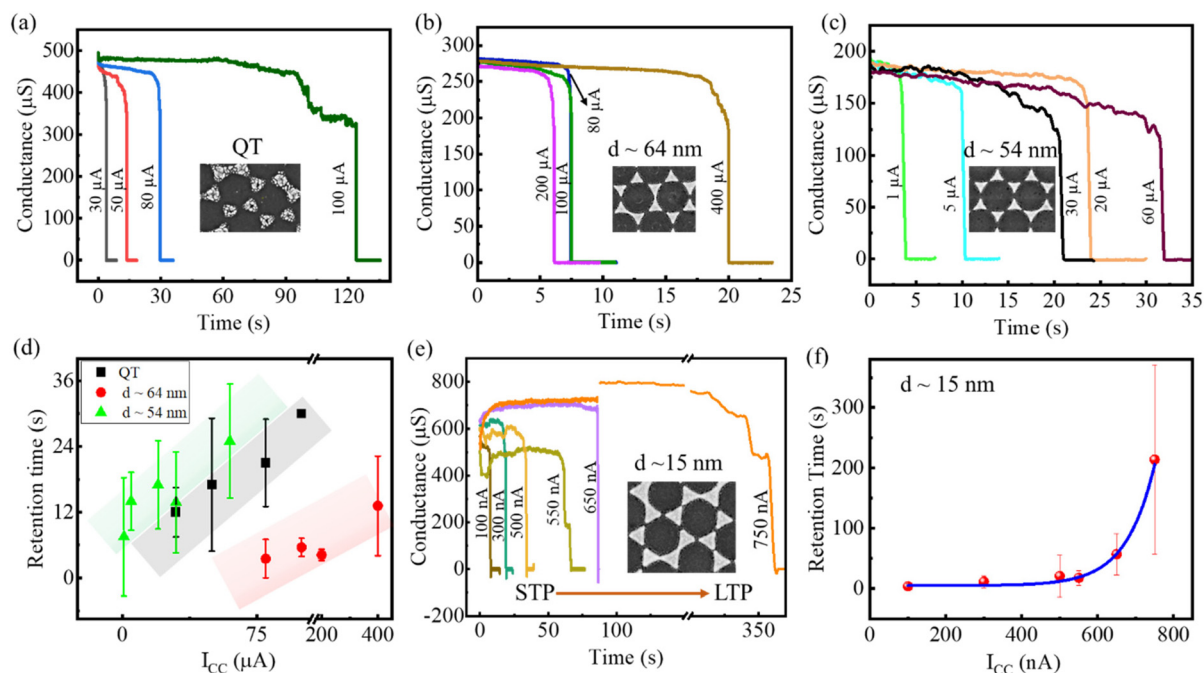


Fig. 4 Memory retention and STP to LTP transition as a function of I_{CC} for devices having (a) QT structure, (b) $d \sim 64$ nm (c) $d \sim 54$ nm. Retention time variation as a function of I_{CC} for (d) QT structure, $d \sim 64$ nm, and $d \sim 54$ nm. (e) STP to LTP transition of conductance at low I_{CC} (nA) of $d \sim 15$ nm device and its (f) retention time versus I_{CC} plot.

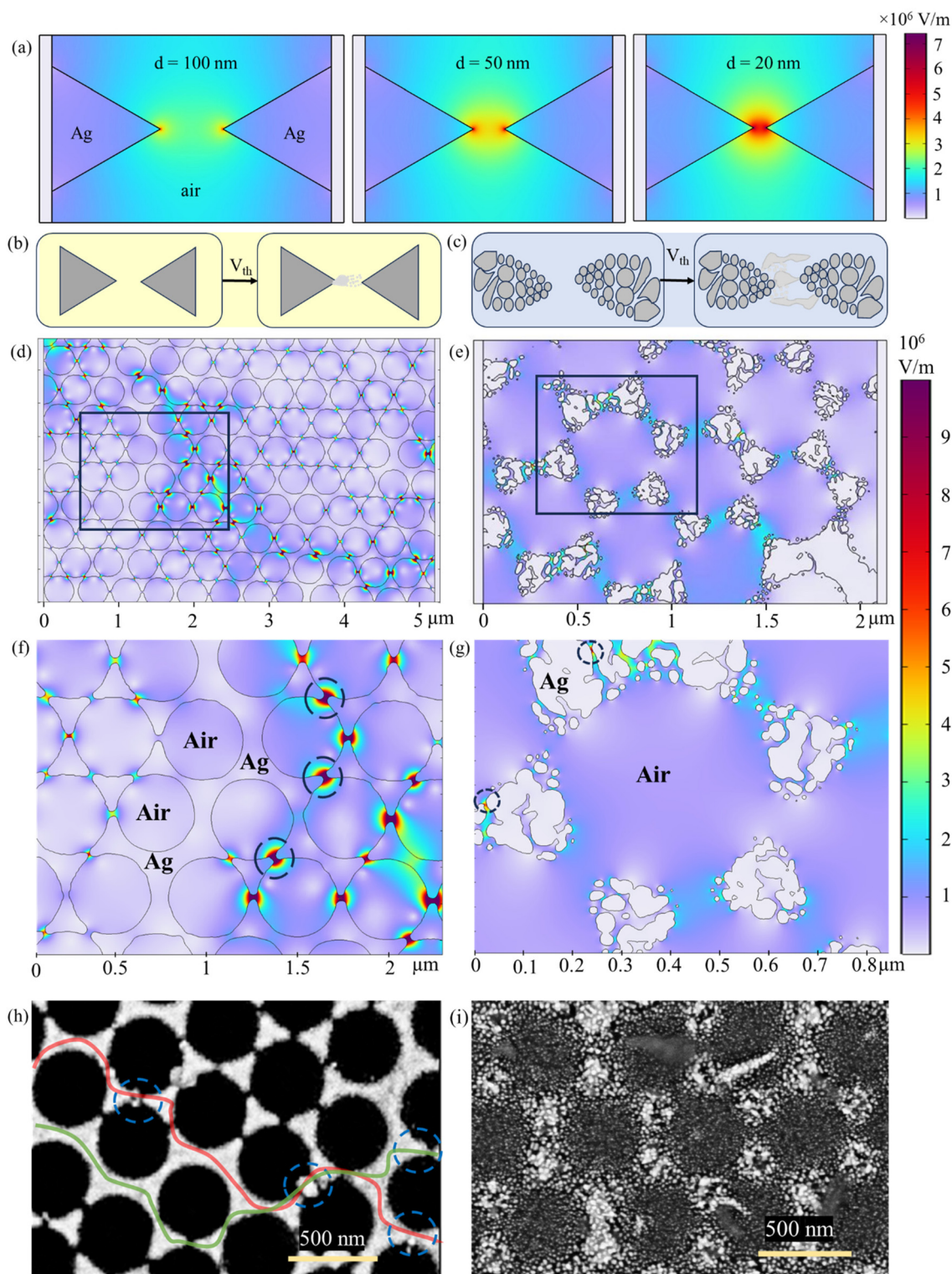


Fig. 6 (a) COMSOL Multiphysics simulation to obtain electric field distribution for Ag bowtie structures with different nanogaps with 0.5 V applied on the left Al electrode. Schematic of threshold switching mechanism of devices with (b) triangular and (c) QT structures. Simulated electric field distribution of (d) Ag triangular nanostructure and (e) QT structure. (f) The magnified image of the marked area in (d), shows the hot spots depicting possible paths for filament formation driven by the maximum field intensity, and (g) the marked region in (e), shows the maximum field distribution within the Ag nanoclusters (blue dashed circle). Post-pulsing SEM image showing (h) traces of filaments (red line – path 1, and green line – path 2) across nanogaps (blue dashed circles) of triangular Ag structures and (i) electromigration in random directions of QT structure, which deteriorates its original structure. Note: voltages of 0.5 and 1 V were applied to devices with triangular and QT nanostructure, respectively.

I_{CC} , the retention time increased with nominal stochasticity (as shown in Fig. 4d), indicating a system apt for probabilistic computing with a high error tolerance.⁵³ Furthermore, the rates of retention increment for $d \sim 54$ (green) and $d \sim 64$ nm (red) infer that STP to LTP transition is more favorable for lower gap devices. The same has also culminated from repeated pulsing measurements at different I_{CC} (Fig. S11–S13†). Applying high electrical stress to a device can result in either the formation of fewer filaments with larger diameters or multiple weaker filaments. In the first scenario, further increasing the current compliance (I_{CC}) enhances retention, leading to LTP. In the second scenario, however, the increased Joule heating from the high current causes the dissolution of the weaker filaments due to their limited current-carrying capacity, producing STP behavior.³¹ Notably, the device with the lowest d requires a minimum voltage of 0.8 V (conductance read by low voltage $V_R = 500 \mu\text{V}$) and only 5 pulses with $t_w = 50$ and $t_i = 50$ ms sufficient to train the device. The device shows an STP to LTP transition with increased I_{CC} (Fig. 4e), and a further enhanced memory formation with iterated training at each I_{CC} (Fig. S14†). It is worth noting that a device with $d \sim 15$ nm requires a much lower I_{CC} for retention compared to other devices, depicting minimal energy consumption. Fig. 4f shows that the retention time exponentially increases with increase in I_{CC} , which could be cognitively correlated with attentive learning.⁵⁴ The overall I_{CC} dependency may be understood from strengthening Ag filaments with the increasing I_{CC} leading to higher retention.

The cognitive learning aspect of the device was demonstrated by emulating Pavlov's classical conditioning or associative learning, as shown in Fig. 5. In associative learning, the brain exhibits a response to a conditioned stimulus (such as a bell) after training involving repetitive association with an unconditioned stimulus (such as food). To emulate such behavior, the device was exposed to two different sets of pulses having an amplitude of 1.2 V (food) and 0.5 V (bell). Initially, a response was observed for the food; however, no response was observed for the bell pulse, as expected. During the training, food and bell pulse were applied concurrently to establish an association between the two signals. The device starts responding to bell pulse after the training. Creating an association between two unlinked stimuli shows the potential of the device to emulate numerous cognitive behaviors of the brain.

For a mechanistic insight and to comprehend the energy consumption trend observed previously, a COMSOL Multiphysics simulation was performed across different nanogaps to determine the electric field intensity and its distribution (see Fig. 6a). It was observed that the decreasing gap between the two nanotriangle confines and enhances the local electric field at the sharp edges, thereby reducing the required voltage for electromigration.⁵⁵ The field distribution helps to identify the probable mechanism and the nanoscopic phenomena, as illustrated schematically in Fig. 6b and c for the bowtie structure and QT, respectively. The triangle structures (Fig. 6b) show predictable and stable filament formations only across the nanogaps where the electric field is confined.

Furthermore, the structure remains stable under continuous pulsing, thereby increasing the durability of the devices. Contrarily, filaments in the QT structure (Fig. 6c) suffer from large stochasticity due to the dislocation of Ag nanoparticles throughout the substrate, effectively increasing the V_{th} over the training period. For the further visualization of a larger picture in the filamentary path formations, COMSOL simulation (details in Note S2†) was performed at a larger scale, as seen in Fig. 6d for the bowtie structure. The simulation predicts that filaments tend to grow across the nanogap where the field is greater than the electric discharge of air, *i.e.*, 3 MV m^{-1} ,⁵⁶ and the magnified image (Fig. 6f) of the marked area in Fig. 6d shows the location of the hotspots. On the other hand, the electric field distribution of the QT structure (Fig. 6e) indicates the lower field intensity between nanogaps than air electric discharge ($<3 \text{ MV m}^{-1}$) and the magnified image (Fig. 6g), confirming that the field is high between the Ag nanoclusters (blue dashed circle), leading to random electromigration that damages the original structure. Simulation results suggest that in triangular structures, the confinement of the electric field within the nanogaps facilitates directional electromigration. The insight has been empirically verified by FESEM imaging (see Fig. 6h and i), where the device was first taken to a high conducting state through pulsing, followed by imaging. As indicated by the blue circles in Fig. 6h, filaments grow only across the gaps to form complete traces of percolation (EDS mapping and spectrum shown in Fig. S15a–d†), as shown by the red and green lines. The post-pulsing FESEM image of the QT structure suffers disoriented filament formation, which causes the diffusion of particles all over the surface, as shown in Fig. 6i.

Thus, the present investigation illustrates an approach to desirably confine filaments in planar neuromorphic devices featuring a hierarchical network structure, enhancing performance and achieve computing with minimal energy consumption. The prospective study will focus on investigating the plasmonic effect in conjunction with neuromorphic properties while concurrently exploring the feasibility of utilizing both light and electrical stimuli.

Conclusions

In summary, we have successfully fabricated a planar neuromorphic device with a periodic Ag nanostructure featuring a controllable gap achieved through plasma assisted NSL. Perhaps this would be the first instance of using NSL for neuromorphic device fabrication. For a comparative analysis, we obtained I - V characteristics for multiple devices with varying gaps controlled *via* OPT for different durations. The device with the minimum gap of 15 nm demonstrates an ultra-low V_{th} of ~ 300 mV, resulting in minimal energy consumption of 8.4 fJ per synapse, comparable to that of a biological synapse. Moreover, the memory and learning behavior of these devices were showcased through appropriate pulsing stimuli, mimicking biological synapses. Notably, the device with the smallest

gap exhibits a STP to LTP transition below the microampere range, outperforming many reported literature cases, thanks to the unique geometry of the Ag nanostructure obtained *via* NSL. Additionally, the psychological behavior of the brain was successfully emulated in an associative learning experiment by employing appropriate pulsing to associate a conditioned (bell) stimulus with an unconditioned one (food). Mechanistically, COMSOL simulation revealed an electric field enhancement across the nanogap between the triangular structures, which confines filament formation between the gaps, thereby reducing energy consumption. This insight was empirically verified through FESEM imaging, illustrating traces of the percolation path aligning with the direction of the electric field. Thus, the present study makes a significant contribution toward low power-consuming neuromorphic devices by utilizing a hierarchical network geometry featuring plasmonic nanostructures.

Author contributions

Shubham K. Mehta: investigation, methodology, formal analysis, software, writing – original draft; Indrajit Mondal: conceptualization, investigation, validation, data curation, writing – review and editing; Bhupesh Yadav: investigation, methodology, writing – review and editing; Giridhar U. Kulkarni: conceptualization, data curation, formal analysis, project administration, resources, supervision, writing – review & editing. All authors have approved the final version of the manuscript.

Data availability

The data supporting this article have been included as part of the ESI.†

Conflicts of interest

There are no conflicts to declare.

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References

- G. Lee, J. H. Baek, F. Ren, S. J. Pearton, G. H. Lee and J. Kim, *Small*, 2021, **17**, e2100640.
- S. Jiang, S. Nie, Y. He, R. Liu, C. Chen and Q. Wan, *Mater. Today Nano*, 2019, **8**, 100059.
- X. B. Yin, Z. H. Tan, R. Yang and X. Guo, *J. Electroceram.*, 2017, **39**, 210–222.
- T. Yu, Y. Fang, X. Chen, M. Liu, D. Wang, S. Liu, W. Lei, H. Jiang, S. Shafie, M. N. Mohtar, L. Pan and Z. Zhao, *Mater. Horiz.*, 2023, **10**, 2181–2190.
- D. Xie, Y. Li, J. He and J. Jiang, *Sci. China Mater.*, 2023, **66**, 4814–4824.
- Z. Huang, Y. Li, Y. Zhang, J. Chen, J. He and J. Jiang, *Int. J. Extreme Manuf.*, 2024, **6**, 032003.
- S. La Barbera, D. Vuillaume and F. Alibart, *ACS Nano*, 2015, **9**, 941–949.
- T. Masquelier, R. Guyonneau and S. J. Thorpe, *Neural Comput.*, 2009, **21**, 1259–1276.
- M. Prezioso, M. R. Mahmoodi, F. M. Bayat, H. Nili, H. Kim, A. Vincent and D. B. Strukov, *Nat. Commun.*, 2018, **9**, 5311.
- A. Kurenkov, S. Fukami and H. Ohno, *J. Appl. Phys.*, 2020, **128**, 010902.
- J. Park, H. Ryu and S. Kim, *Sci. Rep.*, 2021, **11**, 16601.
- Z. Wang, S. Joshi, S. E. Savel'ev, H. Jiang, R. Midya, P. Lin, M. Hu, N. Ge, J. P. Strachan, Z. Li, Q. Wu, M. Barnell, G. L. Li, H. L. Xin, R. S. Williams, Q. Xia and J. J. Yang, *Nat. Mater.*, 2017, **16**, 101–108.
- J. Li, H. Abbas, D. S. Ang, A. Ali and X. Ju, *Nanoscale Horiz.*, 2023, **8**, 1456–1484.
- T. Shi, R. Wang, Z. Wu, Y. Sun, J. An and Q. Liu, *Small Struct.*, 2021, **2**, 2000109.
- R. Yang, *Chin. Phys. B*, 2020, **29**, 097305.
- M. Li, H. Liu, R. Zhao, F. S. Yang, M. Chen, Y. Zhuo, C. Zhou, H. Wang, Y. F. Lin and J. J. Yang, *Nat. Electron.*, 2023, **6**, 491–505.
- D. P. Sahu and S. N. Jammalamadaka, *Sci. Rep.*, 2017, **7**, 17224.
- Y. Sun, J. Li, S. Li, Y. Jiang, E. Wan, J. Zhang, Y. Shi and L. Pan, *Chip*, 2023, **2**, 100031.
- S. Oh, H. Kim, S. E. Kim, M. H. Kim, H. L. Park and S. H. Lee, *Adv. Intell. Syst.*, 2023, **5**, 2200272.
- G. Milano, M. Aono, L. Boarino, U. Celano, T. Hasegawa, M. Koziacki, S. Majumdar, M. Menghini, E. Miranda, C. Ricciardi, S. Tappertzhofen, K. Terabe and I. Valov, *Adv. Mater.*, 2022, **34**, 2201248.
- K. D. M. Rao, A. A. Sagade, R. John, T. Pradeep and G. U. Kulkarni, *Adv. Electron. Mater.*, 2016, **2**, 1500286.
- Q. Xia, M. D. Pickett, J. J. Yang, X. Li, W. Wu, G. Medeiros-Ribeiro and R. S. Williams, *Adv. Funct. Mater.*, 2011, **21**, 2660–2665.
- M. Kawakita, K. Okabe and T. Kimura, *Appl. Phys. Lett.*, 2016, **108**, 023101.
- S. K. Bose, J. B. Mallinson, R. M. Gazoni and S. A. Brown, *IEEE Trans. Electron Devices*, 2017, **64**, 5194–5201.
- K. Sakai, T. Sato, S. Tani, M. Ito, M. Yagi and J. I. Shirakashi, *AIP Adv.*, 2019, **9**, 055317.
- S. K. Bose, S. Shirai, J. B. Mallinson and S. A. Brown, *Faraday Discuss.*, 2019, **213**, 471–485.
- A. Gee, A. H. Jaafar and N. T. Kemp, *Nanotechnology*, 2020, **31**, 155203.

- 28 S. Liu, X. Zhong, Y. Li, B. Guo, Z. He, Z. Wu, S. Liu, Y. Guo, X. Shi, W. Chen, H. Duan, J. Zeng and G. Liu, *Adv. Sci.*, 2023, **35**, 2212118.
- 29 S. Liu, J. Zeng, Z. Wu, H. Hu, A. Xu, X. Huang, W. Chen, Q. Chen, Z. Yu, Y. Zhao, R. Wang, T. Han, C. Li, P. Gao, H. Kim, S. J. Baik, R. Zhang, Z. Zhang, P. Zhou and G. Liu, *Nat. Commun.*, 2023, **14**, 7655.
- 30 B. Bannur and G. U. Kulkarni, *Mater. Horiz.*, 2020, **7**, 2970–2977.
- 31 R. Attri, I. Mondal, B. Yadav, G. U. Kulkarni and C. N. R. Rao, *Mater. Horiz.*, 2023, **11**, 737–746.
- 32 B. Bannur, B. Yadav and G. U. Kulkarni, *ACS Appl. Electron. Mater.*, 2022, **4**, 1552–1557.
- 33 T. S. Rao, I. Mondal, B. Bannur and G. U. Kulkarni, *Discover Nano*, 2023, **18**, 124.
- 34 B. Yadav, I. Mondal, B. Bannur and G. U. Kulkarni, *Nanotechnology*, 2024, **35**, 015205.
- 35 D. Xie, G. Gao, B. Tian, Z. Shu, H. Duan, W. W. Zhao, J. He and J. Jiang, *Adv. Mater.*, 2023, **35**, 2212118.
- 36 N. Ilyas, J. Wang, C. Li, D. Li, H. Fu, D. Gu, X. Jiang, F. Liu, Y. Jiang and W. Li, *Adv. Funct. Mater.*, 2022, **32**, 2110976.
- 37 C. L. Haynes and R. P. Van Duyne, *J. Phys. Chem. B*, 2001, **105**, 5599–5611.
- 38 M. Domonkos and A. Kromka, *Symmetry*, 2022, **14**, 2642.
- 39 P. Colson, C. Henrist and R. Cloots, *J. Nanomater.*, 2013, **2013**, 948510.
- 40 X. Zhang, C. R. Yonzon and R. P. Van Duyne, *J. Mater. Res.*, 2006, **21**, 1083–1092.
- 41 L. Li, F. Du, X. Zong, L. Cui and Y. Liu, *Appl. Opt.*, 2022, **61**, 6924.
- 42 M. Pisco, F. Galeotti, G. Quero, G. Grisci, A. Micco, L. V. Mercaldo, P. D. Veneri, A. Cutolo and A. Cusano, *Light: Sci. Appl.*, 2017, **6**, 16229.
- 43 T. Qiu, E. M. Akinoglu, B. Luo, M. Konarova, J. H. Yun, I. R. Gentle and L. Wang, *Adv. Mater.*, 2022, **34**, e2103842.
- 44 S. Huang, Q. Yang, C. Zhang, L. Kong, S. Li and J. Kang, *Thin Solid Films*, 2013, **536**, 136–141.
- 45 D. Ji, T. Li and H. Fuchs, *Adv. Electron. Mater.*, 2017, **3**, 1600348.
- 46 A. A. Osipov, A. E. Gagaeva, A. B. Speshilova, E. V. Endiarova, P. G. Bespalova, A. A. Osipov, I. A. Belyanov, K. S. Tyurikov, I. A. Tyurikova and S. E. Alexandrov, *Sci. Rep.*, 2023, **13**, 3350.
- 47 Y. Chen, D. Shi, Y. Chen, X. Chen, J. Gao, N. Zhao and C. P. Wong, *Nanomaterials*, 2019, **9**, 605.
- 48 Y. C. Chang, S. M. Wang, H. C. Chung, C. Bin Tseng and S. H. Chang, *Plasmonics*, 2011, **6**, 599–604.
- 49 K. Schraml, M. Spiegl, M. Kammerlocher, G. Bracher, J. Bartl, T. Campbell, J. J. Finley and M. Kaniber, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2014, **90**, 035435.
- 50 J. Tang, F. Yuan, X. Shen, Z. Wang, M. Rao, Y. He, Y. Sun, X. Li, W. Zhang, Y. Li, B. Gao, H. Qian, G. Bi, S. Song, J. J. Yang and H. Wu, *Adv. Mater.*, 2019, **31**, 1902761.
- 51 D. P. Sahu, K. Park, J. Han and T. S. Yoon, *APL Mater.*, 2022, **10**, 051111.
- 52 F. Ahmed, M. S. Choi, X. Liu and W. J. Yoo, *Nanoscale*, 2015, **7**, 9222–9228.
- 53 S. Misra, L. C. Bland, S. G. Cardwell, J. A. C. Incorvia, C. D. James, A. D. Kent, C. D. Schuman, J. D. Smith and J. B. Aimone, *Adv. Mater.*, 2023, **35**, 2204569.
- 54 S. T. Keene, A. Melianas, Y. van de Burgt and A. Salleo, *Adv. Electron. Mater.*, 2019, **5**, 1800686.
- 55 H. Masuda and T. Kizuka, *J. Nanosci. Nanotechnol.*, 2014, **14**, 2436–2440.
- 56 M. F. Abd Alameer and T. H. Khalaf, *IOP Conf. Ser.: Mater. Sci. Eng.*, 2020, **757**, 012018.