

Cite this: *Nanoscale*, 2024, **16**, 17886

Alternating vs. random amphiphilic polydisulfides: aggregation, enzyme activity inhibition and redox-responsive guest release†

Sukanya Bera and Suhrit Ghosh *

Herein, we report the synthesis of an alternating copolymer (ACP) with a bio-reducible amphiphilic polydisulfide backbone and highlight the impact of the alternating monomer connectivity on the self-assembly, morphology, chain-exchange dynamics, drug-release kinetics, and enzyme activity inhibition. Condensation polymerization between hydrophobic 1,10-bis(pyridin-2-ylsulfaneyl)decane and hydrophilic 2,3-mercaptopropanoic acid (1.04 : 1.00 ratio) generated amphiphilic ACP P1 ($M_w = 8450 \text{ g mol}^{-1}$, $D = 1.3$), which exhibited self-assembly in water, leading to the formation of an ultra-thin (height $< 5.0 \text{ nm}$) entangled fibrillar network. In contrast, structurally similar amphiphilic random copolymer P2 exhibited a truncated irregular disc-like morphology under the same conditions. It is postulated that due to the perfect alternating sequence of the hydrophobic and hydrophilic segments in P1, its immiscibility-driven aggregation in water leads to a pleated structure, which further assembles and forms the observed long fibrillar structures, similar to crystallization-driven self-assembly. In fact, wide-angle X-ray diffraction (WAXRD) analysis of a lyophilized P1 sample showed sharp peaks, indicating its crystalline nature (approximately 37% crystallinity), and these were completely missing for P2. The effect of such distinct self-assembly on the chain-exchange dynamics was probed by fluorescence resonance energy transfer (FRET) using 3,3'-dioctadecylxocarbocyanine perchlorate (DiO) and 1,1'-dioctadecyl-3,3,3',3'-tetramethylindocarbocyanine perchlorate (DiI) as the FRET-donor and -acceptor, respectively. For DiI- and DiO-entrapped solutions of P1, when mixed, no prominent FRET appeared even after 24 h. In sharp contrast, for P2, intense FRET emission occurred, and the FRET ratio (approximately 0.9) reached saturation in approximately 15 h, indicating the greatly enhanced kinetic stability of P1 aggregates. Glutathione-induced release of encapsulated Nile red showed much slower kinetics for P1 compared to that of P2, which was corroborated by the observed slow chain-exchange dynamics of the highly stable alternating copolymer assembly. Furthermore, the well-ordered assembly of P1 exhibited an excellent surface-functional group display (zeta potential of -32 mV compared to -14 mV for P2), which resulted in the effective recognition of the α -chymotrypsin (Cht) protein surface by electrostatic interaction. Consequently, P1 significantly ($> 70\%$) suppressed the enzymatic activity of Cht, while in the presence of P2, the enzyme was still active with $> 70\%$ efficacy.

Received 17th June 2024,
Accepted 25th August 2024
DOI: 10.1039/d4nr02494j

rsc.li/nanoscale

Introduction

Amphiphilic polymers exhibit immiscibility-driven aggregation in water, leading to different nanostructures.¹ They are of interest for different applications, including drug delivery, antibacterial materials, tissue engineering, protein delivery,

and others.² Most of these systems are block copolymers,^{1,2} with only a relatively small number of reports on amphiphilic homopolymers,³ or random⁴ or alternating copolymers (ACPs).⁵ ACPs are unique in the sense that there is a defined sequence of the two different monomers⁶ in this structure. Because of this unique sequence, alternating copolymers exhibit distinct self-assembly behaviour⁷ or functional properties.⁸

The first report on the synthesis of an ACP of stilbene with maleic anhydride was published in 1930.⁹ However, it continues to be a challenge to prepare structurally diverse copolymers with an alternating sequence through chain-growth mechanisms. Steric hindrance, pseudoconnectivity, increased

School of Applied and Interdisciplinary Sciences, Indian Association for the Cultivation of Science, 2A and 2B Raja S. C. Mullick Road, Kolkata, India 700032.

E-mail: psusg2@iacs.res.in

† Electronic supplementary information (ESI) available: Synthesis of the polymers, materials and methods, additional self-assembly data and experimental detail. See DOI: <https://doi.org/10.1039/d4nr02494j>

difference in electron density, or a pre-defined sequence between the two comonomers are strategies that have resulted in limited success for synthesis of ACPs by chain polymerization.⁵ However, in step-growth polymerization between AA- and BB-type monomers, the resulting polymer intrinsically contains the alternating sequence.

Hence, if these monomers can be designed in such a way that one is hydrophobic and the other is hydrophilic, it will produce amphiphilic ACPs without any ambiguity in their sequence. Similar strategies have been used for the synthesis of different amphiphilic ACPs that exhibit chain-folding regulated self-assembly or crystallization.^{10,11} We envisaged that such well-defined self-assembled systems may be highly relevant for biological applications, for which it is imperative to achieve control over several important parameters such as functional group display, chain-exchange dynamics, or stability.

We recently developed a synthetic methodology for producing polydisulfides (PDSs) through a condensation polymerization pathway.¹² In a sense, it is a unique polymerization technique, in which polymerization of any dithiol (AA) can proceed with commercially available 2,2'-dipyridyl disulfide (BB) by thiol-disulfide exchange reaction. Although it is an AA + BB-type polymerization, the same polydisulfides are produced that would have been achieved by oxidative polymerization of the dithiol monomer.

We are particularly interested in disulfide-containing polymers because they present a bio-reducible linker,¹³ which can be degraded in the presence of glutathione (GSH), especially in an intracellular location due to its significantly higher concentration.¹⁴ We realized that this methodology can be easily extended to prepare amphiphilic ACPs with a degradable PDS backbone, which would enable testing of the relevance of such perfectly sequenced polymers for aqueous self-assembly and biological applications. With this objective, in the current study, we synthesized an amphiphilic ACP with a PDS backbone (P1, Scheme 1) and compared its self-assembly, chain-exchange dynamics, and enzyme activity inhibition with a structurally similar random copolymer (P2, Scheme 1).

Results and discussion

Scheme 1 shows the synthesis of P1 and P2. Monomer M1 was synthesized from commercially available 1,10-decanedithiol by

reacting it with excess 2,2'-dipyridyl disulfide and isolating a pale yellow oil in 58% yield. The condensation polymerization reaction at rt between 1,10-bis(pyridin-2-yl)disulfaneyl)decane (M1, hydrophobic) and commercially available 2,3-dimercaptosuccinic acid (M2, hydrophilic) produced the desired poly(disulfide)-based amphiphilic ACP P1. It was isolated as a colourless solid in 42% yield. During the polymerization, M1 was intentionally used in slight excess (M1 : M2 = 1.04 : 1.00) to avoid the presence of reactive thiol groups at the chain terminal.

P1 was characterized by ¹H NMR (Fig. 1) in which all the peaks were unambiguously assigned. An enlarged view of the selected region ($\delta = 7.0$ – 8.0 ppm) showed the presence of characteristic peaks for the aromatic protons of the pyridyl-disulfide groups, present in the chain-ends. Also visible was the peak at $\delta = 13$ ppm for the carboxylic acid protons. By comparing the intensity of the end-group protons (He) and backbone proton (Ha), the average degree of polymerization could be calculated to be 17. The UV/Vis spectrum (Fig. S1†) showed a band ($\lambda = 250$ – 300 nm) corresponding to the end-groups, from which the molecular weight of P1 was estimated to be 8560 g mol^{-1} . This was a very good match with that estimated from

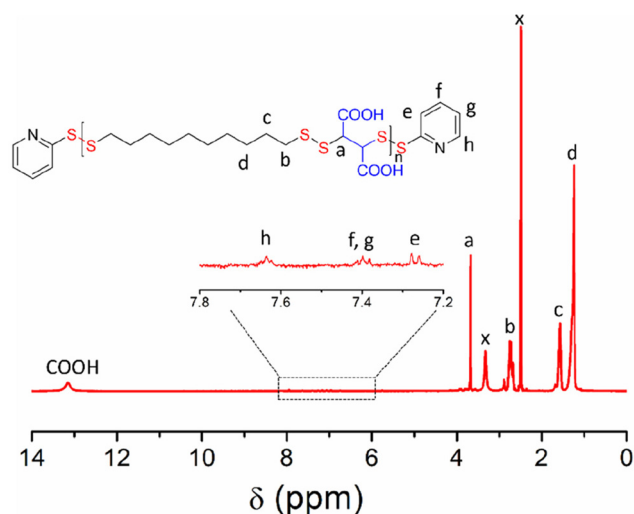
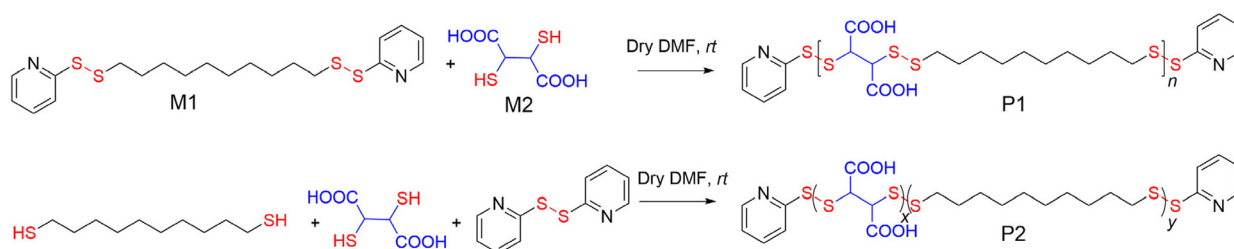


Fig. 1 ¹H NMR of P1 (solvent DMSO-*d*₆); the peaks marked with an 'x' were derived from the solvent.



Scheme 1 Synthesis of P1 and P2.

Table 1 Molecular weight of P1 and P2. Theoretically estimated molecular weight value = 9870 g mol⁻¹ using the stoichiometric imbalance and conversion 100%

Polymer	M_w (SEC) ^a g mol ⁻¹	Dispersity (D)	Mol. wt (NMR)	Mol. wt (UV)
P1	8500	1.25	6800	8600
P2	8700	1.21	7200	8300

^aSEC conditions: concentration = 5.0 mg mL⁻¹; solvent = DMF; standard = polymethyl methacrylate (PMMA).

size-exclusion chromatography (SEC) (Fig. S2†) or end-group analysis using the NMR spectrum (Table 1).

Likewise, the amphiphilic random copolymer P2 was synthesized by condensation polymerization between 1,10-decanedithiol, 2,3-dimercaptosuccinic acid (M2), and 2,2'-dipyridyl disulfide, in which the reactive pyridyl-disulfide and the thiol groups were taken in a 1.04 : 1.00 ratio, while the two dithiols were taken in a 1 : 1 ratio. It produced an amphiphilic random copolymer, which was characterized by SEC studies (Fig. S2†), ¹H NMR (Fig. S3†), and UV/Vis (Fig. S1†).

The molecular weight of P2, estimated from SEC or end-group analysis, was comparable to that of P1 (Table 1). From the end-group analysis by ¹H NMR, the average number of hydrophobic and hydrophilic units was 9 and 10, respectively. It is imperative to note that the molecular weight and dispersity values for both polymers are nearly identical, and therefore, their comparison is free from any intrinsic structural effects other than the monomer connectivity sequence. Also, the dispersity values in the range of approximately 1.2 for polymers synthesized by the step-growth route are appreciable and indicate well-defined polymerization.

The aggregation of P1 was tested in aqueous medium at a basic pH to ensure that the carboxylic acid groups remain as carboxylates. An atomic force microscopy (AFM) image (Fig. 2a) shows long cylindrical structures with an interconnected network formation.¹⁵ Interestingly, the height of these worm-like structures is <5.0 nm, indicating that these are extremely thin flat structures. Contrary to P1, the random copolymer P2 shows an irregular disc-like morphology (Fig. 2b) with a height <2 nm, indicating significant difference in the aggregation behaviour of the alternating and random amphiphilic copolymers.

It was proposed that due to the strictly alternating sequences of the hydrophilic and hydrophobic segments in P1, immiscibility-driven aggregation in water leads to a pleated structure (Fig. 3a), which further assembles to the observed long fibrillar structures,¹⁵ similar to those observed for the crystallization-driven assembly.¹⁶ In fact, wide-angle X-ray diffraction (WAXRD) analysis of the sample, obtained by lyophilization of the aqueous solution of P1, showed (Fig. 3b) sharp peaks at $2\theta = 12.8, 25.6, 32.2,$ and 39.0° , indicating the crystalline nature of the aggregate.¹⁷ In sharp contrast, no such peaks were noted for random copolymer P2 (Fig. 3b), which highlights the importance of the alternating sequence for the observed well-defined aggregation.

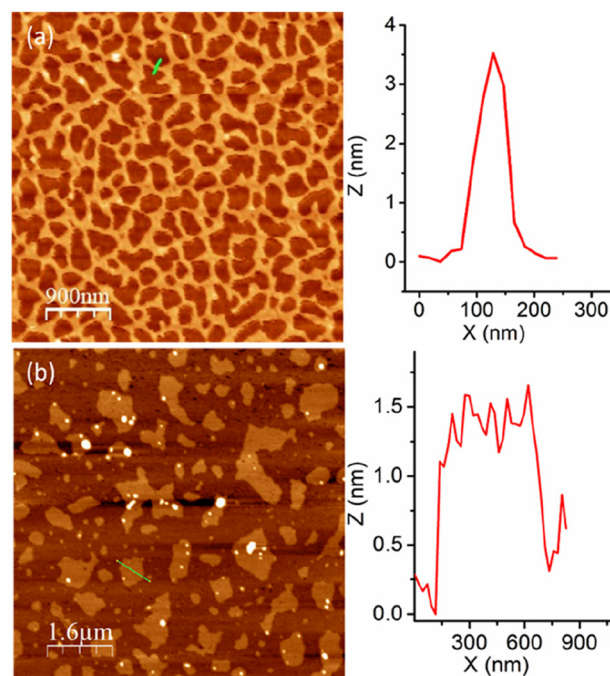


Fig. 2 AFM images of (a) P1 and (b) P2 aggregates. Height profile along the green line is shown at the right.

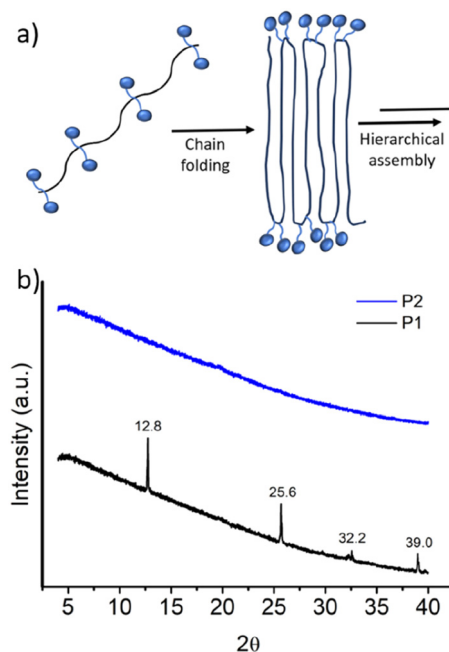


Fig. 3 (a) Schematic showing the immiscibility-driven assembly of P1, with its crystalline nature. (b) Wide angle X-ray diffraction pattern of P1 and P2. Experiments were carried out with solid samples obtained after lyophilization of aqueous aggregates ($c = 3.0$ mg mL⁻¹) of P1 and P2.

The container property of these polymer aggregates was examined by encapsulation of the hydrophobic dye Nile red (NR). In both cases, NR-treated polymer solution showed an intense red colour, indicating successful dye encapsulation in

the hydrophobic pocket of these polymer aggregates. The fluorescence spectra showed a typical emission band for NR. By concentration-dependent fluorescence experiments (Fig. S4†), the critical aggregation concentration (CAC) of P1 and P2 was estimated to be roughly $15 \mu\text{g mL}^{-1}$ and $27 \mu\text{g mL}^{-1}$, respectively.

Next, we examined the chain-exchange dynamics of the aggregates of P1 and P2 by fluorescence resonance energy transfer (FRET) using 3,3'-dioctadecyloxycarbocyanine perchlorate (DiO) and 1,1'-dioctadecyl-3,3,3',3'-tetramethylindocarbocyanine perchlorate (DiI) as the FRET-donor and -acceptor, respectively (Fig. S5†).¹⁸ DiI and DiO were separately encapsulated in the P1 aggregate, and then, the two different dye-encapsulated solutions were mixed together, and the FRET emission was monitored as a function of time (Fig. 4).

Excitation was set at 485 nm, which corresponded to the absorption of the donor (DiO), and resulted in intense emission at 508 nm corresponding to DiO. However, no such prominent emission band was noted for the acceptor DiI at 570 nm, indicating negligible FRET. Emission spectra were recorded up to 24 h, when no increase in the FRET efficiency was noted. Rather, it remained constant at a very low value of approximately 0.3, which can also be due to the contribution from the direct excitation of the acceptor chromophore. This indicates remarkably slow exchange dynamics in the P1 aggregate. In sharp contrast, for the P2 aggregate, after mixing the DiO and DiI encapsulated solutions, the FRET efficiency spontaneously reached a very high value of approximately 0.7

(Fig. 4), which gradually increased and reached a value of >0.9 in 24 h, suggesting a much faster chain exchange in this case.

The contrast, as shown in Fig. 4b, is truly noteworthy because it clearly indicates that the alternating sequence leads to a stable assembly with a negligible chain exchange, which can be attributed to the crystalline nature of the assembly. The irregular sequence in P2 leads to relatively ineffective chain-packing and consequently loosely bonded aggregates and fast chain exchange. It is noteworthy that for applications such as drug delivery and others, it is imperative to have systems with slow exchange dynamics to minimize leakage of the encapsulated therapeutic molecules. This has generally been achieved by crosslinking and other techniques.¹⁹ In that sense, the present system is unique because it demonstrates that crystalline assembly intrinsically can lead to a situation where the chain exchange is extremely slow.

Polydisulfides are of interest because the disulfide linkage can be cleaved in the presence of GSH, which is of biological relevance due to its significantly higher intra-cellular concentration compared to its extracellular domain. GSH is a highly polar tripeptide, while the disulfide linkage in the aggregates is located in the hydrophobic domain. Hence, cleavage of the disulfide bond by diffusion of the GSH to the hydrophobic pocket of the aggregates may not be very effective. Instead, earlier studies with disulfide-containing small molecule surfactants predicted that cleavage may occur in the unimer state of the surfactant, which is always in dynamic equilibrium with the aggregates.²⁰ In that case, the sharp contrast in the chain-exchange dynamics between P1 and P2 may significantly influence the kinetics of GSH-triggered disulfide cleavage and disassembly.

To test such possibilities, the Nile red-encapsulated P1 and P2 aggregates were treated with GSH (10 mM), and the emission spectra of Nile red were monitored as a function of time (Fig. S6,† Fig. 5). The Nile red emission intensity was comparable in both samples.

However, with time, the emission intensity decreased at a much faster rate for P2, and eventually it became negligible after approximately 5 h. For P1, the intensity decreased at a much slower rate, and after the same time interval, significant band intensity remained. The % of dye release was estimated using the band intensity at 622 nm, and was plotted in Fig. 5b. A clear difference was noted, with a much faster release rate for P2, where approximately 90% of the dye was released after 5 h. In the case of P1, a much slower release rate was noted, with only approximately 50% dye release after 5 h. A similar trend was observed when dye release experiments were performed using UV/Vis absorption spectroscopy (Fig. S7†).

This clearly indicates a remarkable influence of the chain-exchange dynamics on the rate of disassembly of these polydisulfide aggregates, which can be clearly corroborated by earlier reports suggesting that such disulfide cleavage occurs in the unimer state of the surfactant rather than in the aggregated state. It is imperative to note that depending on the specific need, both fast release and sustained release are important. Hence, the ability to tune the release rate to such an extent by controlling the polymer sequence is not only of fundamental

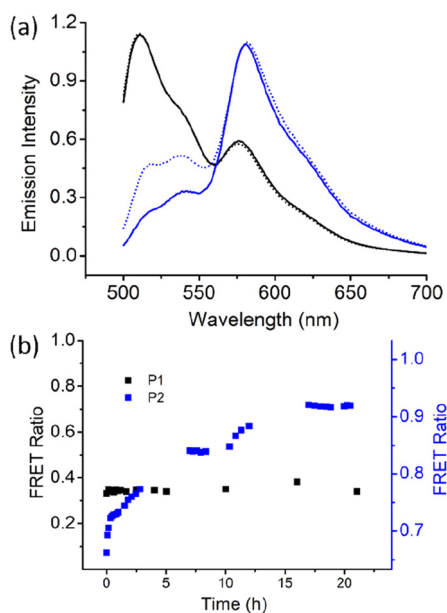


Fig. 4 (a) Emission spectra ($\lambda_{\text{ex}} = 485 \text{ nm}$) of the mixed solution of DiO- and DiI-encapsulated P1 (black line) and P2 (blue line) just after mixing (dotted line) and after 24 h (solid line). For P1 and P2, peak intensities were normalized to 1.0 at 520 nm and 575 nm, respectively. The concentration of polymer and each dye = $20 \mu\text{M}$ and $10 \mu\text{M}$, respectively. (b) Variation of the FRET ratio as a function of time.

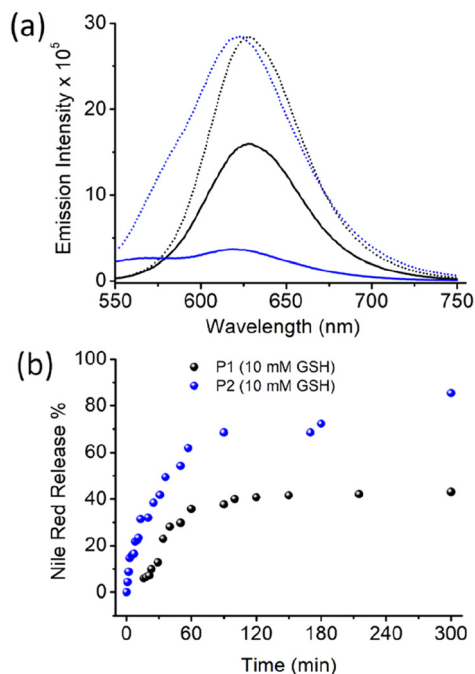


Fig. 5 (a) The emission spectra of Nile red encapsulated in P1 (black line) or P2 (blue line) aggregates before (dotted line) and after GSH treatment for 5 h (solid line). The concentrations of Nile red, P1/P2, and GSH were 10^{-5} M, 0.5 mg mL^{-1} , and 10 mM, respectively. $\lambda_{\text{ex}} = 530 \text{ nm}$. (b) Cumulative % release of Nile red as a function of time after GSH addition. The release % was calculated using an emission intensity of 622 nm.

interest, but also could be of importance for practical applications.

Finally, we examined the effect of the well-ordered assembly of P1 on the surface-functional group display and attempted to correlate it with recognition of the protein surface by electrostatic interaction. For this study, α -chymotrypsin (Cht) with a cationic surface charge was used as the model protein.

The polymers are negatively charged due to the presence of the carboxylate groups, and therefore, it is anticipated that they will bind to the protein surface by electrostatic interaction, which in turn will hamper the enzymatic activity of Cht, depending on the effectiveness of a given polymer aggregate to bind to the protein surface.²¹ To test this, activity assays were carried out with a chromogenic substance, namely *N*-succinyl-L-phenylalanine-*p*-nitroanilide (SPNA) (Fig. 6).

It is known that Cht, in its active form, can hydrolyse SPNA, which produces *p*-nitroaniline with an absorption band in the visible region ($\lambda_{\text{max}} = 405 \text{ nm}$). The activity of Cht in the absence of any polymer was initially confirmed, and a sharp increase in the absorption intensity at 405 nm with time was observed, which indicated that the protein was in its active form. Cht, preincubated with P1, showed negligible production of *p*-nitroaniline, indicating a prominent decrease in the enzymatic activity by approximately 70%. However, in the presence of P2, the decrease in enzymatic activity was only 30%, which was significantly less than that of P1.

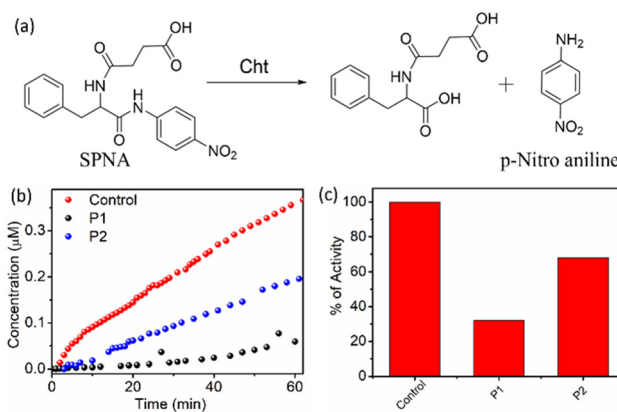


Fig. 6 (a) Scheme showing the Cht-induced product formation of the SPNA substrate. (b) Time versus concentration plot of *p*-nitroaniline generated from the hydrolysis of SPNA (2 mM) in the presence of Cht ($3.2 \mu\text{M}$) after incubation with P1 or P2 (0.1 mM) or without addition of polymer (control) at pH 9.0. (c) Relative Cht activity in the presence of P1 and P2.

Therefore, it is evident that the strict alternating sequence in P1 not only leads to stable self-assembly, but also assists in well-defined chain packing, which leads to a more optimal functional group display. This was also evident from a significantly higher negative zeta potential (-32 mV) for P1 compared to -14 mV for P2. Finally, HeLa cells were subjected to the MTT assay (Fig. S8†) with P1 and P2 (concentration up to $400 \mu\text{g mL}^{-1}$), which showed that $>85\%$ of cells were alive even after 48 h. This indicated excellent compatibility of cells with the polymers and the possibility of their future biomedical application.

Conclusions

Herein, we described an amphiphilic polydisulfide with an alternating sequence of hydrophobic and hydrophilic monomers, and demonstrated its facile synthesis by condensation polymerization involving the thiol-activated disulfide exchange reaction. Self-assembly studies showed that such an alternating sequence is highly useful to achieve distinct morphology and negligible chain-exchange dynamics, which can be attributed to the immiscibility-driven crystalline chain-packing that is missing in structurally similar random copolymers.

Kinetic stability of the aggregate assists in slow degradation of the backbone disulfide linker in the presence of GHS, which is highly useful for sustained drug release. Furthermore, well-ordered chain packing leads to a distinct ultrathin fibrillar morphology with excellent functional group (carboxylate) display that enables electrostatic interaction-mediated surface recognition of the enzyme chymotrypsin (Cht), with approximately 70% inhibition of its enzymatic activity in contrast to the random copolymer, which exhibited negligible enzyme activity inhibition.

Despite significant research progress in the self-assembly of amphiphilic polymers and their functional utility, alternating copolymers are relatively uncommon, perhaps due to the difficulty in producing such polymers with a perfect alternating sequence. However, easy access to such polymers can be attained by a condensation polymerization approach, and the results reported in this manuscript should be highly inspiring to explore such systems for diverse functional materials.

Author contributions

SB performed all the experimental work and collected the data. The data analysis and manuscript writing were jointly performed by SB and SG. SG conceptualized the work, supervised the project, and raised research funding.

Data availability

The data supporting this article have been included as part of the ESI.†

Conflicts of interest

There are no conflicts to declare.

Acknowledgements

SB thanks IACS for a research fellowship. SG thanks the Technical Research Centre, IACS for funding.

References

- (a) Y. Mai and A. Eisenberg, *Chem. Soc. Rev.*, 2012, **41**, 5969; (b) P. Theato, B. S. Sumerlin, R. K. O'Reilly and T. H. Epps III, *Chem. Soc. Rev.*, 2013, **42**, 7055; (c) T. H. Epps III and R. K. O'Reilly, *Chem. Sci.*, 2016, **7**, 1674; (d) H. Sun, C. P. Kabb, M. B. Sims and B. S. Sumerlin, *Prog. Polym. Sci.*, 2019, **89**, 61; (e) F. D. Jochum and P. Theato, *Chem. Soc. Rev.*, 2013, **42**, 7468.
- (a) N. Kamaly, B. Yameen, J. Wu and O. C. Farokhzad, *Chem. Rev.*, 2016, **116**, 2602; (b) H. Cabral, K. Miyata, K. Osada and K. Kataoka, *Chem. Rev.*, 2018, **118**, 6844; (c) E. Fleige, M. A. Quadir and R. Haag, *Adv. Drug Delivery Rev.*, 2012, **64**, 866; (d) Q. Zhang, N. R. Ko and J. K. Oh, *Chem. Commun.*, 2012, **48**, 7542; (e) R. Chacko, J. Ventura, J. Zhuang and S. Thayumanavan, *Adv. Drug Delivery Rev.*, 2012, **64**, 836; (f) B. S. Bolu, R. Sanyal and A. Sanyal, *Molecules*, 2018, **23**, 1570; (g) S. R. Mane, A. Sathyan and R. Shunmugam, *ACS Appl. Nano Mater.*, 2020, **3**, 2104; (h) R. Aluri, S. Saxena, D. Joshi and M. Jayakannan, *Biomacromolecules*, 2018, **19**, 2166; (i) S. Saxena and M. Jayakannan, *Biomacromolecules*, 2020, **21**, 171.
- T. S. Kale, A. Klaikherd, B. Popere and S. Thayumanavan, *Langmuir*, 2009, **25**, 9660.
- (a) L. Li, K. Raghupathi, C. Song, P. Prasad and S. Thayumanavan, *Chem. Commun.*, 2014, **50**, 13417; (b) Y. Hirai, T. Terashima, M. Takenaka and M. Sawamoto, *Macromolecules*, 2016, **49**, 5084.
- K. Nishimori and M. Ouchi, *Chem. Commun.*, 2020, **56**, 3473.
- N. Badia and J.-F. Lutz, *Chem. Soc. Rev.*, 2009, **38**, 3383.
- (a) S. G. Fenimore, L. Abezgauz, D. Danino, C. C. Ho and C. C. Co, *Macromolecules*, 2009, **42**, 2702; (b) M. Ueda, A. Hashidzume and T. Sato, *Macromolecules*, 2011, **44**, 2970; (c) K. Nishimori, E. Cazares-Cortes, J. M. Guigner, F. Tournilhac and M. Ouchi, *Polym. Chem.*, 2019, **10**, 2327; (d) B. Saha, N. Choudhury, S. Seal, B. Ruidas and P. De, *Biomacromolecules*, 2019, **20**, 546.
- (a) E. Zhao, J. W. Y. Lam, L. M. Meng, Y. Hong, H. Q. Deng, G. X. Bai, X. H. Huang, J. H. Hao and B. Z. Tang, *Macromolecules*, 2015, **48**, 64; (b) B. Saha, K. Bauri, A. Bag, P. K. Ghorai and P. De, *Polym. Chem.*, 2016, **7**, 6895; (c) J. J. Yan, Z. K. Wang, X. S. Lin, C. Y. Hong, H. J. Liang, C. Y. Pan and Y. Z. You, *Adv. Mater.*, 2012, **24**, 5617; (d) B. Saha, N. Choudhury, A. Bhadrans, K. Bauri and P. De, *Polym. Chem.*, 2019, **10**, 3306.
- T. Wagner-Jauregg, *Ber. Dtsch. Chem. Ges. B*, 1930, **63**, 3213.
- R. Barman, A. Mukherjee, A. Nag, P. Rajdev and S. Ghosh, *Chem. Commun.*, 2023, **59**, 13951.
- V. Damodara, H. Sardana and S. Ramakrishnan, *Eur. Polym. J.*, 2024, 112818.
- D. Basak, R. Kumar and S. Ghosh, *Macromol. Rapid Commun.*, 2014, **35**, 1340.
- (a) R. Bej, P. Dey and S. Ghosh, *Soft Matter*, 2020, **16**, 11; (b) I. Altinbasak, M. Arslan, R. Sanyal and A. Sanyal, *Polym. Chem.*, 2020, **11**, 7603; (c) J. Zhuang, M. Gordon, J. Ventura, L. Li and S. Thayumanavan, *Chem. Soc. Rev.*, 2013, **42**, 7421; (d) J. Quinn, M. Whittaker and T. Davis, *Polym. Chem.*, 2017, **8**, 97; (e) H. Mutlu, E. B. Ceper, X. Li, J. Yang, W. Dong, M. M. Ozmen and P. Theato, *Macromol. Rapid Commun.*, 2019, **40**, 1800650; (f) J.-H. Ryu, S. Jiwanich, R. Chacko, S. Bickerton and S. Thayumanavan, *J. Am. Chem. Soc.*, 2010, **132**, 8246; (g) I. Altinbasak, S. Kocak, R. Sanyal and A. Sanyal, *Biomacromolecules*, 2022, **23**, 3525.
- (a) G. K. Such, Y. Yan, A. P. R. Johnston, S. T. Gunawan and F. Carus, *Adv. Mater.*, 2015, **27**, 2278; (b) A. Russo, W. DeGraff, N. Friedman and J. B. Mitchell, *Cancer Res.*, 1986, **46**, 2845; (c) F. Q. Schafer and G. R. Buettner, *Free Radicals Biol. Med.*, 2001, **30**, 1191.
- (a) J. C. Foster, S. Varlas, B. Couturaud, Z. Coe and R. K. O'Reilly, *J. Am. Chem. Soc.*, 2019, **141**, 2742; (b) A. Sikder, S. Chakraborty, P. Rajdev, P. Dey and S. Ghosh, *Acc. Chem. Res.*, 2021, **54**, 2670.
- (a) X. He, Y. He, M.-S. Hsiao, R. L. Harniman, S. Pearce, M. A. Winnik and I. Manners, *J. Am. Chem. Soc.*, 2017, **139**, 9221; (b) X. He, M. S. Hsiao, C. E. Boott, R. L. Harniman,

- A. Nazemi, X. Li, M. A. Winnik and I. Manners, *Nat. Mater.*, 2017, **16**, 481; (c) M. Inam, G. Cambridge, A. Pitto-Barry, Z. P. L. Laker, N. R. Wilson, R. T. Mathers, A. P. Dove and R. K. O'Reilly, *Chem. Sci.*, 2017, **8**, 4223; (d) P. J. Hurst, A. M. J. Rakowski and P. Patterson, *Nat. Commun.*, 2020, **11**, 4690; (e) A. Rajak and A. Das, *Angew. Chem., Int. Ed.*, 2022, **61**, e202116572.
- 17 C. Zhang, H. Pan and Y. Zhou, *Macromolecules*, 2023, **56**, 7870.
- 18 P. Rajdev and S. Ghosh, *J. Phys. Chem. B*, 2019, **123**, 327.
- 19 (a) J.-H. Ryu, R. T. Chacko, S. Jiwpanich, S. Bickerton, R. P. Babu and S. Thayumanavan, *J. Am. Chem. Soc.*, 2010, **132**, 17227; (b) P. Rajdev, D. Basak and S. Ghosh, *Macromolecules*, 2015, **48**, 3360.
- 20 S. Ghosh, K. Irvine and S. Thayumanavan, *Langmuir*, 2007, **23**, 7916.
- 21 (a) H. S. Park, Q. Lin and A. D. Hamilton, *J. Am. Chem. Soc.*, 1999, **121**, 8; (b) B. S. Sandanaraj, D. R. Vutukuri, J. M. Simard, A. Klaiherd, R. Hong, V. M. Rotello and S. Thayumanavan, *J. Am. Chem. Soc.*, 2005, **127**, 10693; (c) M. De, S. S. Chou and V. P. Dravid, *J. Am. Chem. Soc.*, 2011, **133**, 17524; (d) A. Sikder, A. Das and S. Ghosh, *Angew. Chem., Int. Ed.*, 2015, **54**, 6755; (e) P. Khanra, P. Rajdev and A. Das, *Angew. Chem., Int. Ed.*, 2024, **63**, e202400486.