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Preparation of noble metal-free porous CuO–ceria and zirconia mixed oxide catalysts using the ammonia driven deposition precipitation method for toluene combustion†

Wouter Van Hoey, ^a Anna Rokicińska, ^b Marek Dębosz, ^b Izabela Majewska, ^a Piotr Kuśtrowski ^b and Pegie Cool ^{*a}

Cu-containing $\text{Ce}_x\text{Zr}_{1-x}\text{O}_2$ noble metal-free mixed metal oxide catalysts were obtained using a non-disclosed synthesis protocol that involves a soft template assisted hydrothermal approach for the synthesis of various $\text{Ce}_x\text{Zr}_{1-x}\text{O}_2$ supports and the subsequent deposition of Cu with a water-based ammonia driven deposition precipitation (ADP) method. As shown using N_2 -sorption, X-ray diffraction (XRD), Raman spectroscopy, UV-vis diffuse reflectance spectroscopy (UV-vis-DR) and scanning electron microscopy (SEM), varying the ratio between Ce and Zr resulted in different crystal structures, going from purely cubic CeO_2 to tetragonal $\text{Ce}_{0.25}\text{Zr}_{0.75}\text{O}_2$ and finally to ZrO_2 with co-existing tetragonal and monoclinic crystal phases. Additionally, the introduction of zirconium to the CeO_2 crystal lattice induced the formation of mesopores, yielding a larger available surface for ADP deposition of Cu. The addition of different Cu loadings had an impact on both structural and optical properties of the catalysts, as well as on their reducibility, which was examined via H_2 -temperature programmed reduction (H_2 -TPR). During testing of the catalytic performance in the combustion of toluene as a model reaction for aromatic VOC combustion, all catalysts showed extremely high selectivity towards H_2O and CO_2 and were more active than CuO-modified commercially available CeO_2 and ZrO_2 supports. The best performance was obtained over the Cu– $\text{Ce}_{0.75}\text{Zr}_{0.25}\text{O}_2$ catalysts whereby the different loadings of Cu yielded nearly identical results. The high activity of these catalysts can be attributed to the successful ADP deposition and the increased amount of oxygen vacancies due to structural defects induced by the incorporation of smaller Zr^{4+} cations into the cubic crystal structure of ceria. The presence of oxygen vacancies was analyzed using X-ray photoelectron spectroscopy (XPS).

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1. Introduction

Volatile organic compounds (VOCs), which by definition are organic compounds with a boiling point ≤ 250 °C at atmospheric pressure, are amongst the major contributors to global air pollution. VOCs are emitted from a wide variety of sources and play an important role in aerosol, photochemical smog and ground-level ozone formation, hence contributing to global climate change.^{1,2} Besides their impact on the environment, VOCs also pose a threat towards human health, since many VOCs are carcinogenic, mutagenic and teratogenic.³ Due to

their toxic nature, strict regulations regarding VOC emissions have been implemented. More specifically, the European Union and its member states agreed on a new National Emissions Reduction Commitments (NEC) Directive (2016/2284/EU), which is designed to reduce the health impacts of air pollution, in comparison to 2005, by at least 50% by 2030. In order to reach this ambitious target, efficient techniques for VOC elimination from polluted air, including adsorption, membrane separation, thermal oxidation, and (photo)catalytic combustion, have attracted extensive research.⁴ From these methods, catalytic combustion is believed to be the most promising due to its low cost, low operational temperature (200–500 °C) and limited formation of by-products.⁵

At present, noble metal based catalysts are considered to be the most efficient for VOC abatement.⁶ In particular, Pt (e.g. Pt– Co_3O_4 ,⁷ Pt/Ce– Al_2O_3 ,⁸ Pt/Eu₂O₃– CeO_2 ,⁹) and Pd (e.g. Pd– CeO_2 ,¹⁰ Pd– SiO_2 ,¹¹ Pd– Al_2O_3 ,¹²) based catalysts exhibit very high catalytic activity at relatively low temperatures. However, the high

^a Laboratory of Adsorption and Catalysis, Department of Chemistry, University of Antwerp, Universiteitsplein 1, Wilrijk B-2610, Belgium.

E-mail: pegie.cool@uantwerpen.be

^b Department of Chemical Technology, Jagiellonian University, Gronostajowa 2, Krakow 30-387, Poland

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cost of these materials and the fact that they are prone to deactivation caused by poisoning¹³ or sintering,¹ has led scientific research towards exploring cheaper and more easily accessible transition metal oxide based catalysts. Recent examples of interesting noble-metal-free catalysts include: Ni-doped α -MnO₂,¹⁴ CuO-CeO₂-nanorod,¹⁵ CuMn₂O₄ spinel¹⁶ and CoCe_{0.75}Zr_{0.25}-Ni foam monolith.¹⁷

Among the interesting catalysts regarding the VOC abatement, ceria-based materials have gained a lot of scientific attention.¹⁸ This interest comes from the fact that ceria and its related materials are omnipresent in research concerning oxidative catalytic processes, such as automotive three-way-catalysis,^{19,20} water-gas shift reaction^{21,22} and the preferential oxidation of CO (CO-PROX),^{23,24} due to their high redox activity and exceptional oxygen storage capacity (OSC). The latter is caused by the presence of lattice defects in the crystal structure, which induce oxygen mobility. As a result, besides Langmuir-Hinshelwood (L-H) and Eley-Rideal (E-R), the Mars-van Krevelen (MVK) mechanism, whereby lattice oxygen is involved, can also be activated in oxidation reactions.^{18,25}

According to the literature, the OSC of the widely used ceria can be enhanced significantly by incorporating zirconium ions in the CeO₂ crystal lattice. The smaller size of a Zr⁴⁺ cation (ionic radii of 0.84 Å compared to 0.97 Å for Ce⁴⁺) modifies the cubic fluorite structure of CeO₂, which results in an enhanced OSC and oxygen mobility, improving the redox properties of the material.^{26,27} Moreover, the addition of zirconia also improves the thermal stability of ceria.^{27,28}

In order to further enhance the catalytic performance of Ce/Zr-oxide based materials in the VOC combustion, transition metal oxides, such as CuO,^{29,30} MnO_x,^{31,32} and Co₂O₃,³³ can be added. To date, X-Ce-Zr (X = Cu, Co, Mn) ternary mixed oxide catalysts have already gained tremendous attention regarding other applications, due to their excellent thermal stability and superior redox properties.³⁴ On top of that, it is well known from the literature that the combination of transition metals with Ce/Zr oxides greatly improves the catalytic oxidation of VOCs, due to their enhanced redox properties and enhanced oxygen mobility.³² Zhang *et al.* recently reported that upon deposition on TiO₂ nanoparticles the multi-metal combination of CuCeZr shows the highest toluene removal efficiency and selectivity towards CO₂, followed by MnCeZr, NiCeZr, CoCeZr and FeCeZr, respectively.³⁵ Moreover, it was proven that the oxidation process of toluene is dominated by lattice oxygen present in CuO species dispersed on CuCeZrO_x catalysts.³⁶ Furthermore, apart from its relatively low price and high reusability with a recycled yield up to 99%, Cu is known to have a strong synergistic effect with CeO₂.^{37,38} As a result, Cu-Ce-Zr mixed oxide catalysts have attracted significant scientific interest for the abatement of VOCs. However, it is known that the catalytic activity of transition metal supported ceria-based catalysts strongly depends on transition metal dispersion and its interaction with the support material, which are often determined by the selection of synthesis parameters.^{37,39} Therefore, the presented study reports for the first time in the literature the deposition of Cu onto Ce_xZr_{1-x}O₂

catalysts using the ammonia driven deposition precipitation (ADP) method.

The ADP method, which was first implemented by Guo *et al.*⁴⁰ and further optimized by Xin *et al.*⁴¹ for the improved deposition of Cu onto SBA-15, involves the selective adsorption of a copper tetra-ammonia complex followed by its deposition during drying. Upon drying, the Cu complex is deposited on the support material, followed by a thermal activation step, forming a chemical bond between the Cu-sites and the support surface. Consequently, the ADP method ensures a strong metal-support interaction, high degree of metal dispersion and high stability of the deposited CuO sites.⁴¹ Furthermore, ADP is an aqueous based deposition method, hence environmentally friendly and easy for up-scaling.

Due to the valuable characteristics of the ADP method, it has already been successfully applied for Cu deposition onto TiO₂ and Al₂O₃ supports.^{42,43} However, its application for Cu deposition onto Ce_xZr_{1-x}O₂ catalysts is yet to be reported. Therefore, the aim of the presented study is to analyze the influence of the ADP deposition of Cu onto multiple Ce_xZr_{1-x}O₂ ($x = 0; 0.25; 0.5; 0.75; 1$) supports by studying the resulting properties of the active catalyst phase and the catalytic performance in toluene combustion. Two different Cu loadings are examined, namely 10 and 12.5 wt%. These loadings are deliberately selected as a 10 wt% Cu loading is proven to be effective for VOC combustion and 12.5 wt% is the highest attainable Cu loading that ensures a strong metal support interaction with the utilized supports without a significant loss of the active element upon ADP synthesis.⁵

2. Experimental

Used chemicals: NH₄OH (25%); CeCl₃·7H₂O, 99%, Merck; ZrCl₄, 98%, Sigma-Aldrich; Cu(NO₃)₂·3H₂O, >99%, Merck; *N*-hexadecyl-*N,N,N*-trimethylammonium bromide, CTAB, >99%, Acros; cerium(IV) oxide powder <5 micrometer (99.9% trace metals basis) Sigma-Aldrich; ZrO₂ J.P. Pharma-Chem Industries.

2.1. Catalyst preparation

Porous Ce_xZr_{1-x}O₂ supports, with x equal to 0, 0.25, 0.5, 0.75 or 1, were designed using a template-based approach. 6.0 g CTAB was dissolved in 50 mL of deionized water at 30 °C to speed up the process. To this solution a second solution containing 50 mL of deionized water and either only ZrCl₄ (98%, Sigma-Aldrich) or CeCl₃·7H₂O (99%, Merck) or a mixture of both metal salt precursors in amounts relative to the desired ratio between Zr and Ce was added under vigorous stirring in a dropwise fashion. After the addition the temperature was raised to 50 °C and the resulting mixture was stirred for 30 min. Subsequently, a solution containing 20 mL 25% NH₄OH diluted with 20 mL deionized water was added dropwise. The resulting mixture was stirred overnight at 50 °C and then transferred to a Teflon-lined stainless steel autoclave pressure vessel, where it was kept at 100 °C for 24 h, as advised by Dou *et al.*⁴⁴ Next, the obtained



slurry was filtered, washed with distilled water and dried at room temperature. Finally, the obtained powder was calcined at 450 °C for 5 h at 1 °C min⁻¹.

After the successful synthesis of the desired supports, Cu was deposited using the ammonia driven deposition precipitation approach, whereby, typically 1 g of support was suspended in an appropriate amount of 0.03 M copper nitrate solution to obtain the desired theoretical loading of 10 wt% or 12.5 wt%. Then, ammonia (NH₄OH, 28–30% Acros) was added into the mixture to gain a molar Cu/NH₃ ratio of 1/6. Subsequently, the obtained solution was stirred for 48 h at room temperature, followed by filtration and washing with a sufficient amount of deionized water. After drying at room temperature the obtained solid was calcined at 550 °C for 6 h at a heating rate of 1 °C min⁻¹. The final samples were referred to as y Cu–Ce_xZr_{1-x}O₂, where y represents the desired wt% of Cu.

Finally, commercial zirconia and ceria were used as reference supports, onto which 10 wt% of Cu was deposited according to the same ADP synthesis protocol as described above.

2.2. Catalyst characterization

2.2.1. XRD. In order to determine the crystallographic structure of the materials, X-ray diffraction (XRD) measurements were performed using an X-ray diffractometer (Bruker D8 Advance eco) equipped with CuK α radiation ($\lambda = 0.15406$ nm). The XRD patterns were recorded with a speed of 0.04° s⁻¹ in the 2θ range of 5° to 80°.

2.2.2. EDXRF. Energy dispersive X-ray fluorescence (EDXRF) analysis was used for a more thorough analysis of the chemical composition of the synthesized materials. These measurements were performed using a Thermo Scientific ARL QUANTX EDXRF C10020 X-ray fluorescence spectrometer with a Rh anode and a Si(Li) detector. All spectra were processed using the UniQuant ED tool.

2.2.3. N₂-sorption. Information about the total BET specific surface area and the pore size distribution of the catalysts was obtained using N₂-adsorption–desorption. The measurements were performed at a temperature of –196 °C using a Quantachrome Quadrasorb SI surface area and pore size analyzer. Before any N₂-sorption measurement, the samples were degassed under vacuum during 16 h at a temperature of 200 °C by using an AS-6 degasser.

2.2.4. SEM. Scanning electron microscopy (SEM) micrographs were recorded using an Apreo 2S Thermo Fisher Scientific scanning electron microscope at an accelerating voltage of 10 and 20 kV. All samples were placed on a metallic SEM stub with conductive carbon tape and coated with a 10 nm gold layer in a vacuum coater (Safematic CCU-010 HV).

2.2.5. H₂-TPR. Temperature-programmed reduction with hydrogen (H₂-TPR) was carried out on 50 ± 0.1 mg of a sample, which was then heated from 100 °C to 900 °C in a flowing mixture of 5 vol% H₂ in Ar (total flow rate of 27 mL min⁻¹) at a heating rate of 10 °C min⁻¹. A thermal conductivity detector (TCD) placed directly after a cold trap (–40 ± 3 °C) was used to determine the amount of hydrogen consumed during the

reduction reaction. The TCD signal was calibrated to the reduction of CuO as a reference material.

2.2.6. Raman spectroscopy. The Raman spectra were recorded using a Horiba xPlora plus Raman spectrometer equipped with a 532 nm laser.

2.2.7. UV-vis-DR spectroscopy. UV-vis diffuse reflectance spectroscopy measurements were performed on a Nicolet Evolution 500 UV-vis spectrophotometer equipped with a deuterium lamp, a mercury lamp and a diffuse reflectance DR accessory. The spectra were recorded in the range of 200–800 nm at a scan speed of 120 nm min⁻¹. Dried KBr was used as a reference material and also to dilute the samples to 2 wt%. The resulting absorption data were obtained after Kubelka–Munk conversion of the measured reflectance spectra. To do so, the Kubelka–Munk function, as given in eqn (1), was utilized, whereby $R(\lambda)$, $K(\lambda)$, and $S(\lambda)$ represent the reflectance, the absorption coefficient, and the scattering coefficient, respectively.

$$F(R) = \frac{(1 - R)^2}{2R} = \frac{K}{S} \quad (1)$$

2.2.8. XPS. X-ray photoelectron spectroscopy (XPS) measurements were performed with a Prevac photoelectron spectrometer equipped with a hemispherical VG SCIENTA R3000 analyzer. The spectra were recorded using an Al K α source ($E = 1486.6$ eV) and a low energy electron flood gun FS40A-PS to compensate charge on the surface of nonconductive samples. The pressure in the analysis chamber during the measurements was equal to 5×10^{-9} mbar. The peaks were recorded with a constant pass energy of 100 eV. The scale of the binding energy value was adjusted by the C 1s reference peak at 284.6 eV. The composition and chemical state of the sample surface were investigated on the basis of the areas and binding energies of Cu 2p, Ce 3d, Zr 3d and O 1s photoelectron peaks. Fitting of the high resolution spectra was carried out using Casa XPS software.

2.3. Catalytic tests

The catalytic activity of all samples was tested for total oxidation of toluene. Additionally, 10 wt% Cu was deposited onto commercial ZrO₂ and CeO₂ to serve as a benchmark during the catalytic tests. A fixed bed microreactor system with 8.0 mm diameter was used. The analysis of released products of the reaction was controlled using a Bruker 450 gas chromatograph equipped with two capillary columns (Porapak S and Chromosorb WAW-DMCS), two flame ionization detectors, a thermal conductivity detector and a methanizer. The catalytic reaction temperature was monitored using a thermocouple inserted into the catalytic bed. 100 mg of powder catalyst was placed on quartz wool in a quartz flow microreactor, then degassed at 500 °C for 30 min in air at a flow rate of 100 mL min⁻¹. After that, the reactor was cooled to 100 °C and dosing the toluene concentration of 1000 ppm (diluted by air stream) began through a saturated scrubber filled with a liquid reagent while keeping the vapor pressure. The catalytic results were collected



at 100, 150, 200, 225, 250, 275, 300, 325, 350, 400, 450 and 500 °C. Each step was maintained for about 70 min and then the reactor was heated to higher temperature at a heating ramp of 10 °C min⁻¹.

3. Results and discussion

3.1. Structural and textural properties of the $\text{Ce}_x\text{Zr}_{1-x}\text{O}_2$ supports

In order to highlight the influence of the catalyst composition on the ADP Cu deposition, various $\text{Ce}_x\text{Zr}_{1-x}\text{O}_2$ supports were utilized for the deposition of both 10 and 12.5 wt% of Cu. The usage of different support compositions should lead to varying structural and textural catalyst properties, potentially influencing the distribution of Cu species. To reveal this, N_2 adsorption–desorption measurements were performed to gain insight into the porosity and total BET specific surface area of the materials. All obtained isotherms are depicted in Fig. 1 and 2,

and the information concerning the corresponding textural parameters is presented in Table 1.

From Fig. 1, it is clear that a different type of isotherm can be ascribed to the monometallic CeO_2 and ZrO_2 supports compared to the Ce/Zr mixed oxides. In the case of pure ceria and zirconia, on the one hand, the shape of the adsorption isotherm corresponds with a type II adsorption isotherm, according to the I.U.P.A.C. classification. This type of isotherm represents an unrestricted monolayer–multilayer adsorption, which typically corresponds to non-porous or macroporous materials. For the pure CeO_2 support, a steep increase in the adsorbed volume can be noticed near a relative pressure of 0.90. This indicates the presence of interparticle porosity, which is confirmed by the presence of a type H1 hysteresis loop. Unlike CeO_2 , the adsorption isotherm of ZrO_2 shows an increase in adsorbed volume at lower relative pressures in combination with a type H3 hysteresis loop, indicating the presence of intraparticle porosity. As a result, the obtained specific BET surface area of ZrO_2 is significantly higher. In the case of the $\text{Ce}_x\text{Zr}_{1-x}\text{O}_2$ mixed oxide supports, on the other hand, a type IV(a) isotherm, resembling mesoporous materials, can be assigned. The presence of mesopores is confirmed, on the one hand, by the pore size distribution, determined using the BJH method and presented in Fig. S1 of the ESI,† and, on the other hand, by the fact that a clear capillary condensation is observed between a relative pressure of 0.40 and 0.90. The resulting hysteresis loop in this relative pressure region can be classified as type H4, which is attributed to slit-shaped pores. As a result, it is clear that the bimetallic combination of cerium and zirconium induces mesoporosity.

To check the crystallographic structure of the involved supports, X-ray diffraction patterns were collected, which are shown in Fig. 3. For pure ceria, eight diffraction peaks are observed at $2\theta = 28.5^\circ, 33.0^\circ, 47.4^\circ, 56.3^\circ, 59.1^\circ, 69.4^\circ, 76.8^\circ$ and 78.9° . These peaks can be attributed to the (111), (200), (220), (311), (222), (400), (331) and (420) reflections of the cubic fluorite phase of ceria, respectively. In the case of $\text{Ce}_{0.75}\text{Zr}_{0.25}\text{O}_2$, a similar diffraction pattern is found, which implies that the

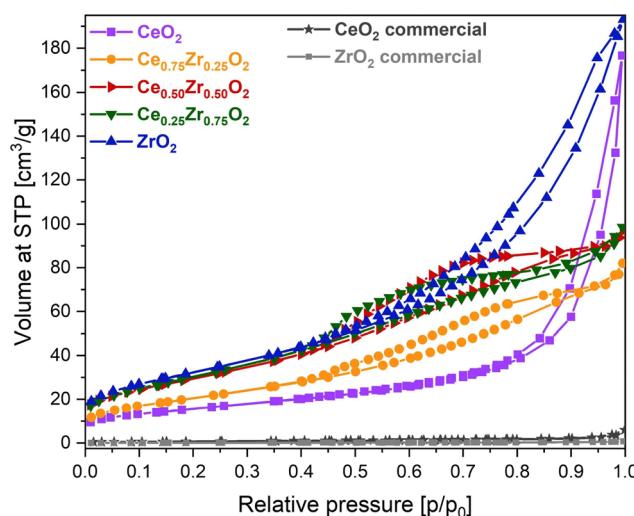


Fig. 1 N_2 -sorption isotherms of the $\text{Ce}_x\text{Zr}_{1-x}\text{O}_2$ supports.

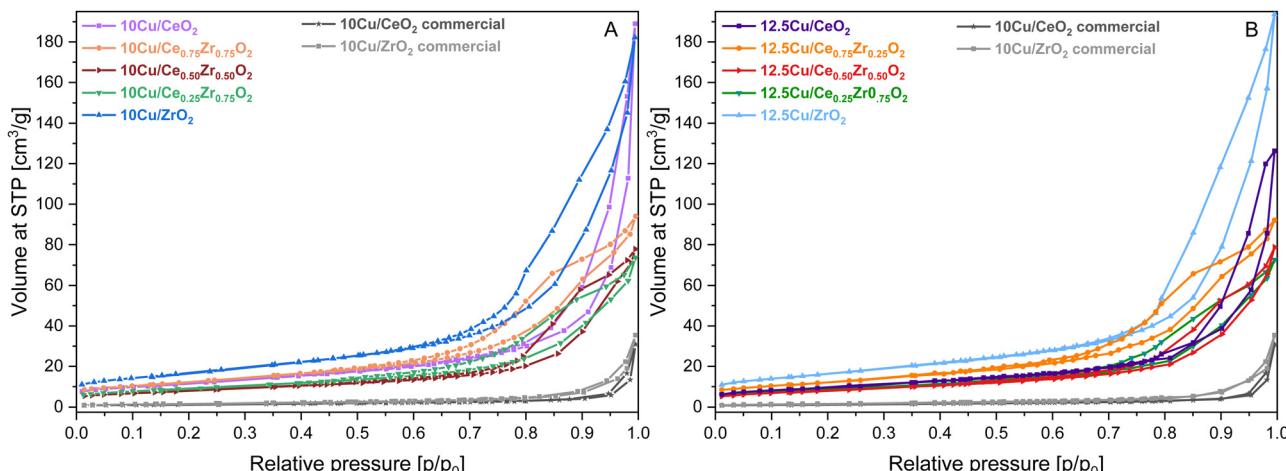
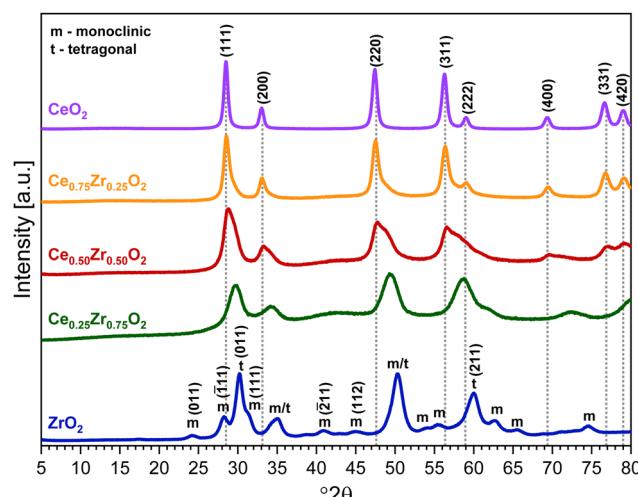


Fig. 2 N_2 -sorption isotherms of the $\text{Ce}_x\text{Zr}_{1-x}\text{O}_2$ supports after ADP deposition of 10 wt% Cu (A) and 12.5 wt% Cu (B).



Table 1 Textural parameters of the studied materials determined by N_2 -sorption measurements

Theoretical Cu-loading	N.A.		10 wt%		12.5 wt%		
	Support material	BET surface area ($m^2 g^{-1}$)	Total pore volume ($cm^3 g^{-1}$)	BET surface area ($m^2 g^{-1}$)	Total pore volume ($cm^3 g^{-1}$)	BET surface area ($m^2 g^{-1}$)	Total pore volume ($cm^3 g^{-1}$)
CeO ₂		55	0.147	41	0.107	34	0.089
Ce _{0.75} Zr _{0.25} O ₂		75	0.112	44	0.118	44	0.117
Ce _{0.50} Zr _{0.50} O ₂		109	0.139	28	0.096	29	0.082
Ce _{0.25} Zr _{0.75} O ₂		113	0.132	32	0.082	30	0.086
ZrO ₂		118	0.250	60	0.180	58	0.188
CeO ₂ -commercial		2	0.004	4.5	0.047	—	—
ZrO ₂ -commercial		0.3	0.001	6	0.055	—	—

Fig. 3 XRD patterns of the $Ce_xZr_{1-x}O_2$ supports.

cubic crystal phase is retained. However, the incorporation of 25% Zr causes a slight shift of the diffraction peaks towards higher Bragg angles. This shift can be explained by the replacement of Ce^{4+} cations (ionic radius of 0.97 Å) with smaller Zr^{4+} (ionic radius of 0.84 Å), which modifies the cubic fluorite structure of CeO_2 .^{26,27} Moreover, the incorporation of Zr also results in peak broadening, which can be ascribed to the formation of smaller crystallite sizes.⁴⁵ The average crystallite size (D) can be estimated by using the well-established Scherrer equation (see eqn (2)), whereby K represents the shape constant, λ is the wavelength of the utilised Cu-K α radiation (0.15406 nm), β is the full width at half maximum of the corresponding diffraction peak expressed in radians, and θ refers to the Bragg angle in radians.

$$D = \frac{K \cdot \lambda}{\beta \cdot \cos \theta} \quad (2)$$

when the diffraction peaks corresponding to the (111), (200) and (220) reflections are taken into consideration, the average size of CeO_2 crystallites for the CeO_2 and $Ce_{0.75}Zr_{0.25}O_2$ support is estimated to be 11.7 and 9.2 nm, respectively, proving the formation of smaller crystallites upon the addition of Zr. Note that these obtained values are merely used for relative comparison, since morphological effects and disorder to reflection

widths are not taken into account. Further peak broadening in combination with a shift towards higher Bragg angles is observed for $Ce_{0.50}Zr_{0.50}O_2$. However, unlike for CeO_2 and $Ce_{0.75}Zr_{0.25}O_2$, it can be noticed that the peaks corresponding with the (220), (311) and (222) reflections are distorted. Hence, it is inferred that a homogeneous ceria–zirconia mixed oxide was not obtained, but that mixed oxides with varying Ce/Zr ratios co-exist.

Looking at the diffractogram of $Ce_{0.25}Zr_{0.75}O_2$, the peaks corresponding to the (400), (331) and (420) reflections are no longer detected. This is caused by the presence of a significantly higher amount of Zr compared to Ce, resulting in the fact that the cubic crystal structure is not retained. To understand this it is important to highlight that apart from a cubic crystal structure, zirconia is known to have two other polymorphs, namely, tetragonal and the thermodynamically stable monoclinic crystal phase. The significant diffraction peaks of $Ce_{0.25}Zr_{0.75}O_2$ can be ascribed to a tetragonal crystal structure, implying that Ce is incorporated into a zirconia lattice. According to Yu *et al.* the presence of lower amounts of cerium in a Ce/Zr mixed oxide stabilizes the tetragonal structure of zirconia.⁴⁵ However, since a broad signal between 38° and 45° and a shoulder between 60° and 65° are found, the presence of a monoclinic ZrO_2 crystal phase cannot be excluded. Finally, in the case of ZrO_2 , both diffraction peaks corresponding with a monoclinic and tetragonal crystal phase are observed. A cubic crystal structure is not obtained, as it is typically only formed at temperatures exceeding 2370 °C.⁴⁶

To support the information derived from the XRD patterns, Raman spectroscopy measurements were performed, which are presented in the ESI.† Additionally, SEM micrographs, which offer an excellent visual examination of the structures formed during the synthesis of the $Ce_xZr_{1-x}O_2$ supports are also presented in the ESI.†

3.2. Characterization of Cu-containing catalysts

3.2.1. Structural and textural properties. After the synthesis of the support materials, both 10 wt% and 12.5 wt% of Cu was deposited using the ADP method. These two different loadings were deliberately selected, because a 10 wt% Cu loading is proven to be effective for VOC combustion and 12.5 wt% is the highest attainable Cu loading that ensures a strong metal support interaction with the utilized supports



Table 2 Cu loadings determined by XRF for the studied materials

Support material	Theoretical loading	
	10 wt% Cu	12.5 wt% Cu
CeO ₂	9.28	12.36
Ce _{0.75} Zr _{0.25} O ₂	7.57	9.57
Ce _{0.50} Zr _{0.50} O ₂	6.41	8.70
Ce _{0.25} Zr _{0.75} O ₂	6.12	7.86
ZrO ₂	6.72	7.60

without a significant loss of the active element upon ADP synthesis.⁵

In order to verify the deposited amount of Cu on the surface of the Ce_xZr_{1-x}O₂ supports, X-ray fluorescence analysis was performed. The collected XRF results are presented in Table 2. From these results it can be noticed that the detected contents of Cu on the CeO₂ support are in both cases very close to the desired amounts. As N₂-adsorption-desorption data proved, CeO₂ is the support with the lowest available surface area, which mainly possesses interparticle porosity. Due to the absence of mesoporosity, Cu can only be deposited on the surface of the material. Hence why the detected amounts of Cu validate the utilized deposition protocol. Furthermore, when the XRF data are analyzed in more detail, it can be noted that typically a lower amount of Cu is detected upon an increasing amount of zirconium. This can be correlated to the total specific surface area of the supports, which is represented in Table 1. More specifically, when the zirconium content increases for the Ce_xZr_{1-x}O₂ supports, a higher surface area is obtained. Due to the increasing porosity and the fact that XRF is considered to be a rather surface technique, it is believed that higher amounts of Cu are potentially situated deeper within the pores and are therefore covered from the interaction volume of the X-rays.

As derived from the N₂-adsorption-desorption measurements presented in Fig. 2 and Table 1, the deposition of Cu and the additional required calcination step resulted in a shift

of the corresponding isotherms towards higher relative pressures and in a significant decrease of the total specific surface area. However, despite the isotherm shift, no changes are to be reported for the classification of the isotherms and hysteresis loops.

The X-ray diffractograms of the Cu-supported materials are depicted in Fig. 4. First of all, due to an additional calcination step involving an increased temperature of 550 °C, a higher degree of crystallinity is obtained for all supports. This also explains the significant decrease in surface area observed during N₂-adsorption-desorption measurements. In accordance with the XRF data, the obtained XRD patterns confirm the successful deposition of CuO on all Ce_xZr_{1-x}O₂ supports. More specifically, the reflections originating from the monoclinic CuO tenorite phase (C2/c space group, ICDD card: 00-005-0661), namely (111), (111), (202), (113), (113), (311) and (004) can be recognized at 35.5°, 38.7°, 48.7°, 61.6°, 66.2°, 68.1°, 72.5° and 75.1° 2θ, respectively. Additionally, for the 12.5 wt% Cu samples, the (020) and (202) reflections can be identified at 53.6° and 58.3° 2θ, respectively.

In the case of ZrO₂ an interesting observation can be made. When the (111) reflection from the monoclinic crystal phase and the (011) reflection from the tetragonal phase are analyzed in more detail for the Cu-containing ZrO₂ materials and the pure ZrO₂ support, it is observed that the diffraction peaks corresponding to the monoclinic crystal phase gain intensity and that those from a tetragonal structure significantly decrease upon an increasing amount of Cu. Consequentially, it can be stated that a higher amount of Cu stabilizes the monoclinic crystal structure, which has been reported in recent studies on similar materials.⁴⁷

When directly comparing the diffractograms of the two different amounts of Cu on the same support, it is clear that a higher amount of Cu results in an increased intensity of the CuO reflections. Furthermore, an increase in intensity of the reflections can be noticed with an increasing amount of zirconium for both the 10 wt% Cu and the 12.5 wt% Cu samples.

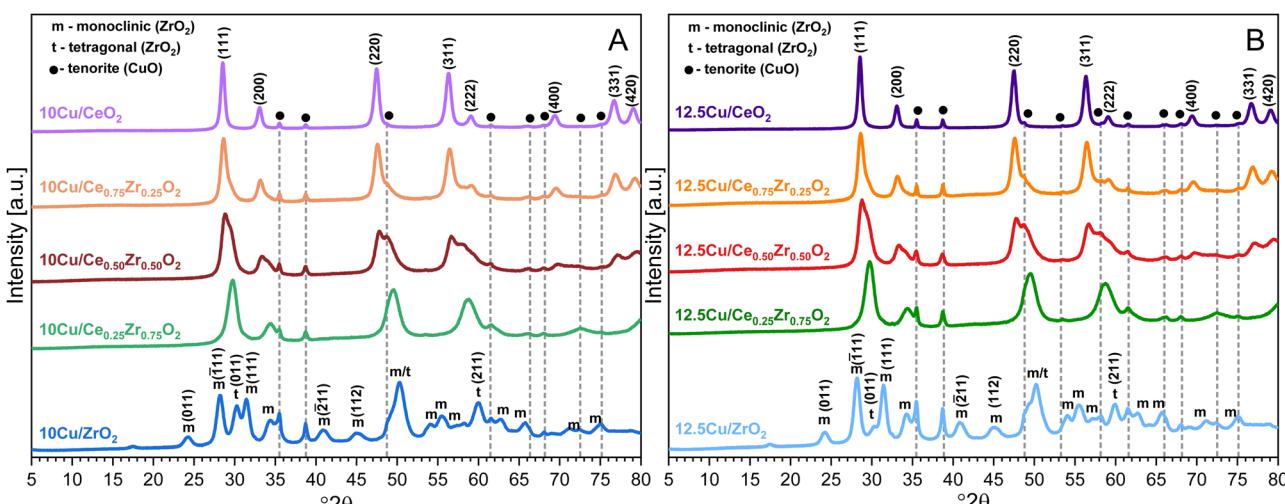
Fig. 4 XRD patterns of the Ce_xZr_{1-x}O₂ supports after ADP deposition of 10 wt% Cu (A) and 12.5 wt% Cu (B).

Table 3 Estimated CuO crystallite size using the Scherrer equation based on the (111) peak of CuO

Support	10 wt% Cu	12.5 wt% Cu
CeO ₂	26.3	33.7
Ce _{0.75} Zr _{0.25} O ₂	28.1	31.2
Ce _{0.50} Zr _{0.50} O ₂	20.5	17.8
Ce _{0.25} Zr _{0.75} O ₂	20.0	22.2
ZrO ₂	21.8	20.0

Utilizing the isolated (111) reflection of CuO, an estimation of the corresponding particle sizes is made through the Scherrer equation. The resulting values are presented in Table 3. From this table, it can be stated that a higher particle size is obtained after deposition of Cu on a support known to possess a lower available surface area. This complies with the idea that larger surface areas should enhance the dispersion of active elements.⁴⁸

In order to support the insights gained through XRD, structural changes and interactions upon the deposition of Cu as the active phase were also studied using Raman spectroscopy. The obtained spectra are presented in Fig. 5. According to the literature, CuO has three Raman active modes, namely one A_g vibration with an estimated position around 290 cm⁻¹ and 2 B_g vibrations positioned around 330 and 620 cm⁻¹.⁴⁹ However, Fig. 5 indicates that the addition of Cu does not result in the presence of characteristic CuO signals but that it leads to a red shift of the Raman peaks in comparison to the pure supports presented in Fig. S2 of the ESI.† Additionally, in the case of CeO₂, Ce_{0.75}Zr_{0.25}O₂ and Ce_{0.50}Zr_{0.50}O₂ the presence of Cu appears to induce the formation of a weak band centered around 270–280 cm⁻¹, which can be attributed to the displacement of oxygen atoms from their ideal positions in a fluorite like crystal lattice.⁵⁰ Moreover, for these three materials the addition of Cu also increases the intensity of the band around 600 cm⁻¹, which corresponds to the longitudinal optical mode

of ceria that arises due to relaxation of symmetry rules, indicating the formation of more oxygen vacancies.⁵⁰

In the case of both Cu–Ce_{0.25}Zr_{0.75}O₂ catalysts, all expected bands corresponding to a tetragonal structure are present. However, in the Raman spectrum of the catalyst containing 12.5 wt% Cu a weak band is observed at 165 cm⁻¹, which can be attributed to the presence of a monoclinic structure. A similar observation is made for the 10Cu/ZrO₂ catalyst where this band is appearing at 176 cm⁻¹. Lastly, when looking at the Raman spectrum of the 12.5Cu/ZrO₂ catalyst in Fig. 5B it can be noticed that the same band referring to a monoclinic structure, which occurs at 165 cm⁻¹, has significantly gained in intensity, in comparison to the 10Cu/ZrO₂ catalyst. This fortifies the statement that a higher loading of Cu tends to stabilize the monoclinic crystal phase.

3.2.2. Optical properties. After gaining insight into how the addition of 10 and 12.5 wt% Cu influenced the porosity and crystal structure of the catalysts, UV-vis diffuse reflectance spectroscopy was utilized to investigate the resulting changes in the optical properties. The resulting Kubelka–Munk transformed diffuse reflectance spectra are presented in Fig. 6.

From Fig. 6 it is clear that the addition of Cu²⁺, yielding CuO after calcination, strongly affects the electronic properties of the pure supports. First of all, it is observed that the absorbance bands of the Cu containing catalysts, with the exception of both Cu–CeO₂ catalysts, show an increased intensity when directly compared to the bands of the corresponding pure support. This can be attributed to the higher degree of crystallinity and the resulting bigger particles obtained after the calcination at an increased temperature of 550 °C, because the given absorption spectra in Fig. 6 were obtained after Kubelka–Munk conversion of the measured reflectance spectra, whereby the scattering coefficient (S) is dependent on the grain size.⁵¹ Consequentially, diffuse reflectance increases with an increasing particle size. In the case of CeO₂, however, it is known that the intensity

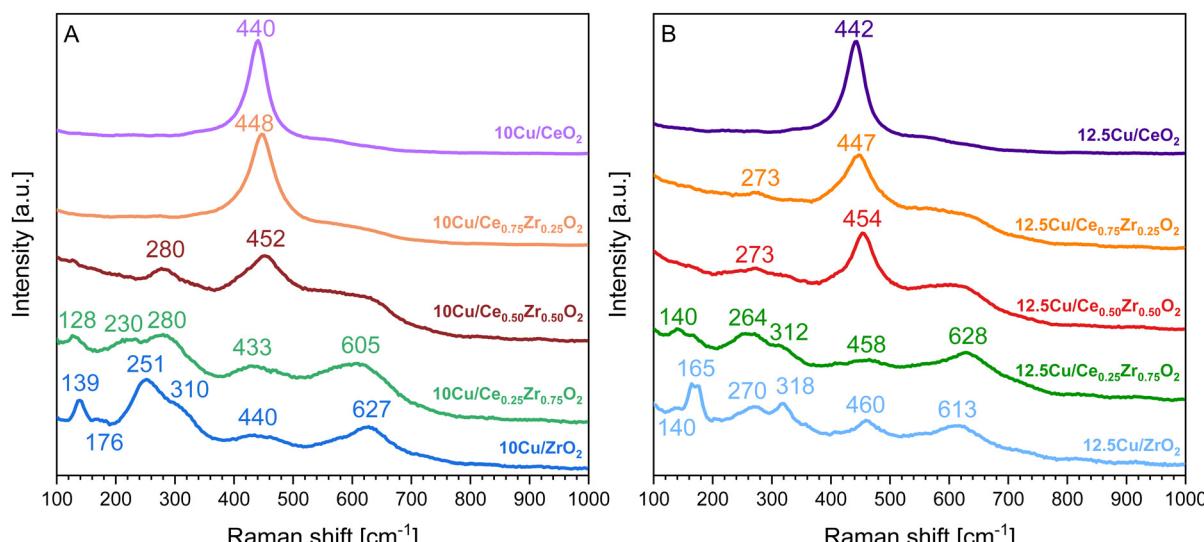


Fig. 5 Raman spectra of the Ce_xZr_{1-x}O₂ supports after ADP deposition of 10 wt% Cu (A) and 12.5 wt% Cu (B).

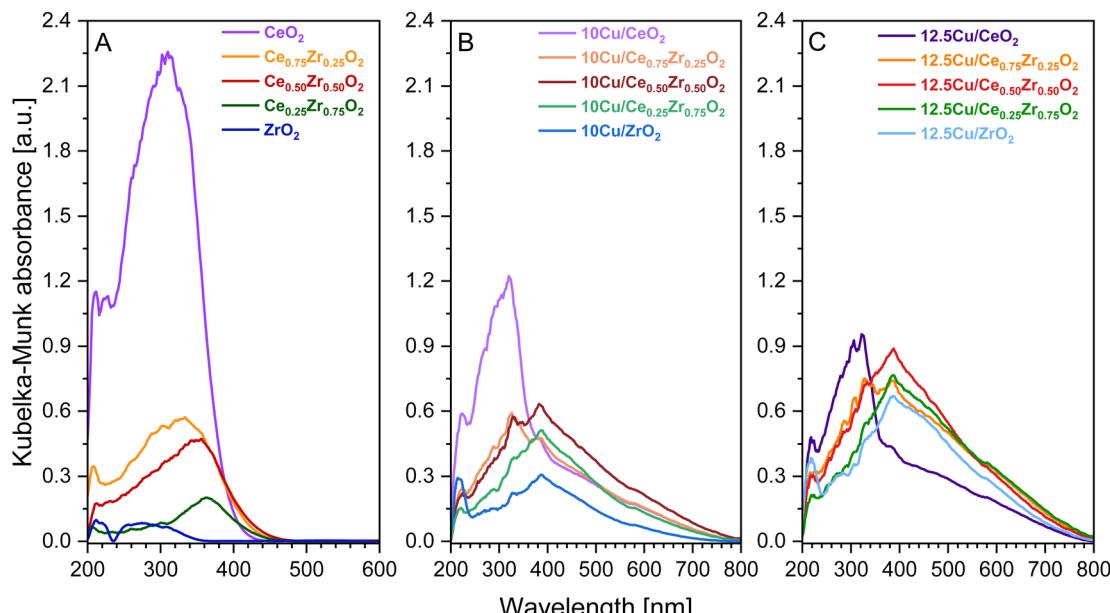


Fig. 6 UV-vis-DR spectra of the pure $\text{Ce}_x\text{Zr}_{1-x}\text{O}_2$ supports (A) and the corresponding 10 wt% Cu (B) and 12.5 wt% Cu (C) catalysts.

of the absorbance gradually decreases with increasing calcination temperatures due to enhanced oxygen atom effusion at higher calcination temperatures generating more oxygen vacancies.⁵² Next, for all Cu-containing catalysts, two absorption maxima related to the copper species can be recognized. The first one is located around 385 nm and can be assigned to the charge transfer from O^{2-} to Cu^{2+} , whereby the Cu-ions occupy isolated positions on the support.^{47,53,54} The strongest absorbance related to this charge transfer is found for the $\text{Cu}-\text{Ce}_{0.50}\text{Zr}_{0.50}\text{O}_2$ catalysts, suggesting the formation of aggregated Cu species and/or bulk CuO. The second maximum is found around 580 nm and indicates the presence of well-dispersed Cu^{2+} ions, which is the result of the successful ammonia driven deposition precipitation of CuO precursor.⁴⁷

When analyzing the obtained spectra of the CeO_2 , $\text{Ce}_{0.75}\text{Zr}_{0.25}\text{O}_2$ and $\text{Ce}_{0.50}\text{Zr}_{0.50}\text{O}_2$ supports in Fig. 6A, which have a defined cubic crystal structure proven by XRD and Raman spectroscopy, two bands can be identified at *ca.* 260 and 300 nm. These bands can be attributed to the charge transfer transition of O^{2-} to Ce^{3+} and O^{2-} to Ce^{4+} , respectively.⁵⁵ Upon introduction of Cu all bands experience a red shift towards higher wavelengths. The same observation can be made after deposition of Cu on $\text{Ce}_{0.25}\text{Zr}_{0.75}\text{O}_2$ and ZrO_2 . Moreover, in the case of $\text{Ce}_{0.25}\text{Zr}_{0.75}\text{O}_2$, the bands corresponding to the charge transfer transition of O^{2-} to Ce^{3+} and O^{2-} to Ce^{4+} , can also be identified. However, unlike the catalysts that contain a cubic crystal structure, $\text{Ce}_{0.25}\text{Zr}_{0.75}\text{O}_2$ is known to have a predominantly tetragonal crystal structure, which results in the presence of a band around 300 nm, resembling the charge transfer transition of O^{2-} to Zr^{4+} .⁵⁶ Finally, when studying the ZrO_2 -based materials, two bands are identified. The first absorption band occurs around 210 nm, which corresponds to the O^{2-} to Zr^{4+} charge transfer

transition in monoclinic- ZrO_2 .^{56,57} The second band arises around 280 nm and reveals the coexistence of tetragonal ZrO_2 , matching the XRD and Raman spectroscopy data.⁵⁶ Additionally, XRD provided evidence that the addition of Cu has a stabilizing effect on the monoclinic crystal phase of ZrO_2 . This is clearly supported by the UV-vis-DR data, as for both 10 and 12.5 wt% Cu-ZrO₂ the absorption band of the O^{2-} to Zr^{4+} charge transfer transition in monoclinic- ZrO_2 , which has redshifted to 220 nm, has significantly gained in intensity compared to the pure ZrO_2 support.

3.2.3. Reducibility. In order to determine the reduction behavior of the catalysts and to understand the interaction between CuO and the $\text{Ce}_x\text{Zr}_{1-x}\text{O}_2$ supports, H₂-temperature programmed reduction measurements were performed. The recorded reduction profiles are presented in Fig. 7. From Fig. 7A it can be derived that the reduction profiles of the Ce-rich supports exhibit two identifiable peaks. The first peak is situated in the range of 500 to 750 °C, and corresponds to the reduction of surface oxygen, whereas the second peak located at higher temperatures is assigned to the reduction of bulk oxygen.^{58–60} The additional small peak for CeO_2 with a maximum around 320 °C can also be associated with the reduction of surface oxygen, with the slight difference that this type of oxygen is less strongly bound to the surface. When looking at the peaks resembling the reduction of surface oxygen, it is observed that their maximum shifts towards higher temperatures upon an increasing amount of zirconium. Furthermore, the bimetallic combination of Ce and Zr significantly boosts the amount of reducible oxygen sites, with $\text{Ce}_{0.75}\text{Zr}_{0.25}\text{O}_2$ showing the highest hydrogen consumption. This indicates that $\text{Ce}_{0.75}\text{Zr}_{0.25}\text{O}_2$ possesses the highest amount of reducible oxygen sites, which are believed to play an important role in the VOC combustion.



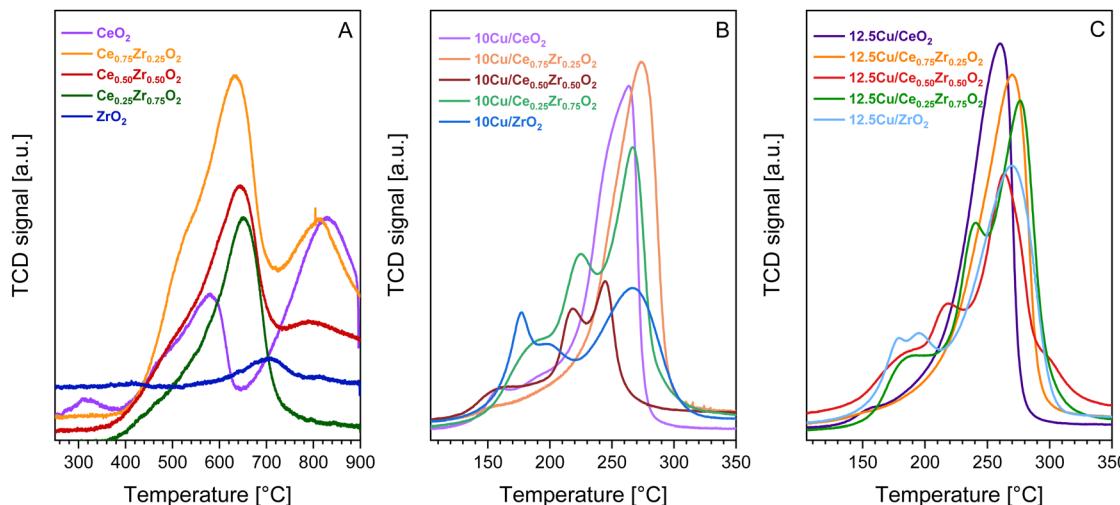


Fig. 7 H_2 -TPR profiles of the pure $\text{Ce}_x\text{Zr}_{1-x}\text{O}_2$ supports (A) and the corresponding 10 wt% Cu (B) and 12.5 wt% Cu (C) catalysts.

When comparing Fig. 7A with B and C, it is clear that the reduction peaks of the supports are located at higher temperatures compared to the reduction peaks of their Cu-containing counterparts. This indicates that the addition of Cu significantly enhances the redox behavior of the studied $\text{Ce}_x\text{Zr}_{1-x}\text{O}_2$ materials. To get a better understanding of how the addition of Cu enhances the reducibility, a more in-depth description of the resulting reduction profiles is presented. Upon analysis of the H_2 -TPR measurements of both the 10 and 12.5 wt% Cu catalysts, varying reduction profiles containing either a single or multiple peaks are observed. This implies that the structural and textural differences of the $\text{Ce}_x\text{Zr}_{1-x}\text{O}_2$ supports impact the resulting Cu active sites. In general, when comparing multiple reduction peaks, the low-temperature reduction peaks typically belong to well dispersed or smaller Cu species, whereas high-temperature signals can be attributed to particles with a larger size, which inhibit weaker metal–support interactions.^{47,61} According to the estimated CuO particle sizes presented in Table 3, the CeO_2 and $\text{Ce}_{0.75}\text{Zr}_{0.25}\text{O}_2$ supports give rise to larger CuO particles, which is believed to be caused by their lower available surface area. As a result, and knowing that both supports possess a clear cubic crystal structure, the corresponding H_2 -TPR profiles exhibit one main signal, indicating a uniform Cu distribution. More specifically, the H_2 -TPR profile of $10\text{Cu}/\text{Ce}_{0.75}\text{Zr}_{0.25}\text{O}_2$ consists of one broad signal centered around 270 °C, which can be attributed to the reduction of bulk CuO and copper species interacting strongly with the support.^{44,62} The signal of $12.5\text{Cu}/\text{Ce}_{0.75}\text{Zr}_{0.25}\text{O}_2$, on the contrary, is broader and more intense, indicating the presence of more active species. Furthermore, both Cu containing CeO_2 materials show similar reduction profiles, with maxima occurring at a slightly lower temperature of around 260 °C. Only one small additional peak is observed around 150 °C. This can be explained by the presence of a low amount of smaller copper particles in close interaction with the ceria lattice.^{44,62,63} For the remaining materials containing 50% or more zirconium, X-ray diffraction analysis indicated the co-existence of multiple

crystal structures. As a result, multiple reduction signals are detected, which implies the formation of different active sites and thus a non-uniform deposition of CuO particles.

3.2.4. Surface composition. The surface electronic states of the as-synthesized 12.5Cu– $\text{Ce}_x\text{Zr}_{1-x}\text{O}_2$ catalysts was studied by using X-ray photoelectron spectroscopy. Fig. 8 shows XPS spectra recorded for the catalysts doped with 12.5 wt% Cu in

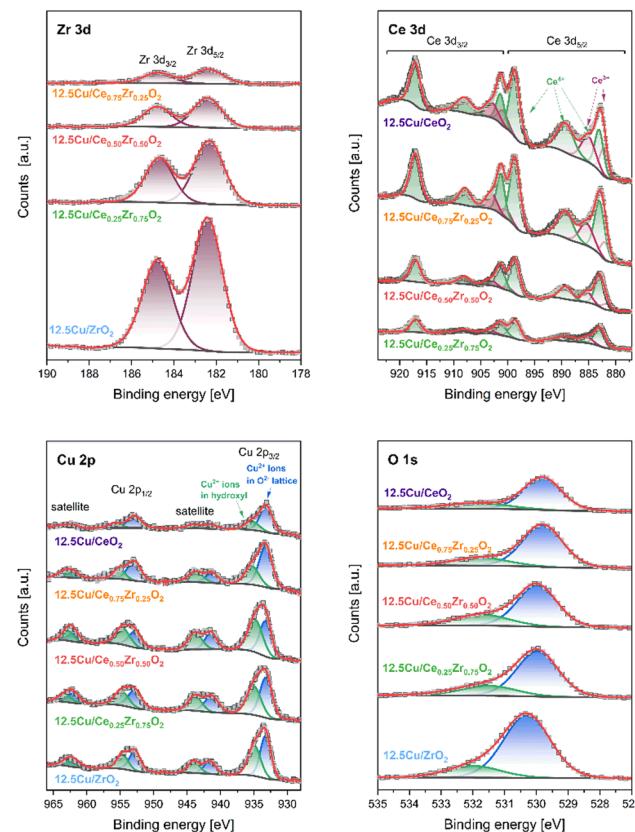


Fig. 8 XPS Zr 3d, Ce 3d, Cu 2p and O 1s spectra of 12.5 wt% Cu catalysts.

Table 4 Surface composition of 12.5 wt% Cu catalysts determined via XPS

Catalyst	Content [at%]				Atomic ratios		
	Zr	Ce	Cu	O	Cu _{hydroxyl} /Cu _{oxide}	Ce ³⁺ /(Ce ³⁺ + Ce ⁴⁺)	O _{latt} /O _{ads}
12.5Cu/ZrO ₂	39.4	—	18.6	42.0	0.82	—	4.61
12.5Cu/Ce _{0.25} Zr _{0.75} O ₂	24.1	6.6	29.7	39.5	0.83	0.13	3.49
12.5Cu/Ce _{0.50} Zr _{0.50} O ₂	12.6	13.1	34.0	40.3	1.18	0.16	2.76
12.5Cu/Ce _{0.75} Zr _{0.25} O ₂	6.4	28.2	27.2	38.1	0.58	0.21	3.84
12.5Cu/CeO ₂	—	39.9	20.3	39.8	0.43	0.28	3.47

areas characteristic of photoelectron emission from Zr 3d, Ce 3d, Cu 2p and O 1s core levels.

In the XPS Zr 3d spectra two components at 182.4 eV and 184.8 eV, corresponding to spin-orbit splitting of 3d_{5/2} and 3d_{3/2} states ($\Delta = 2.4$ eV), confirm the presence of Zr⁴⁺ ions in ZrO₂ lattice.⁵ The emission of photoelectrons from Ce atoms is much more complex. The obtained XPS Ce 3d spectra were fitted with 6 peaks at 883.0 eV (v), 889.3 eV (v'), 898.6 eV (v'''), 901.3 eV (u), 908.0 eV (u''), and 917.1 eV (u''') assigned to Ce⁴⁺ ions (components of 3d_{5/2} are denoted as v, whereas 3d_{3/2} as u).⁶⁴ Furthermore, spin-orbit splitting between Ce 3d_{5/2} and Ce 3d_{3/2} in Ce³⁺ ions results in two additional doublets at 881.9/900.5 eV and 885.4/903.3 eV. In the XPS Cu 2p_{3/2} region two peaks at 933.2 eV and 935.1 eV followed by intense satellite features, related to Cu²⁺ ions present in O²⁻ (Cu_{oxide}) and OH⁻ (Cu_{hydroxyl}) species, are observed.⁶⁵ The occurrence of various surface oxygen forms – oxide lattice (O_{latt}) and hydroxyls (O_{ads}) – is additionally confirmed in the XPS O 1s spectrum, where photoelectron emission is found at 530.0 eV (O_{latt}) and 531.7 eV (O_{ads}).

Table 4 shows several quantitative results of XPS surface analysis for the studied Cu-containing catalysts. Elemental analysis indicates expected changes in the relative content of Zr and Ce while maintaining a similar O concentration (38.1–42.0 at%). However, attention should be paid to significant discrepancies in the amount of Cu species deposited on the

outermost surface. The highest concentration of these centers is achieved for the 12.5Cu/Ce_{0.50}Zr_{0.50}O₂ material. Changing the composition of the support to increase the content of one of the components (CeO₂ or ZrO₂) leads to a decrease in the presence of surface Cu. The discussed feature is additionally correlated with the type of Cu forms. The higher the concentration of Cu on the surface, the higher the Cu_{hydroxyl}/Cu_{oxide} ratio is observed. At the same time, the value of the O_{latt}/O_{ads} ratio decreases. The analysis of the surface composition reveals another interesting effect. The degree of reduction of the CeO₂ lattice increases with increasing Ce content.

3.3. Catalytic performance of Cu-containing catalysts in toluene combustion

The catalytic performance of the synthesized Cu-supported Ce_xZr_{1-x}O₂ materials was tested in the catalytic total oxidation of toluene. Toluene combustion is chosen because it serves as a model reaction for the combustion of aromatic VOCs. The observed toluene conversions in relation to the reaction temperature are presented in Fig. 9. Hereby, it is worth mentioning that for all synthesized catalysts only the desired reaction products, namely CO₂ and H₂O, are detected, since no noticeable signals corresponding to the formation of by-products, such as CO or benzene, are found. This highlights the

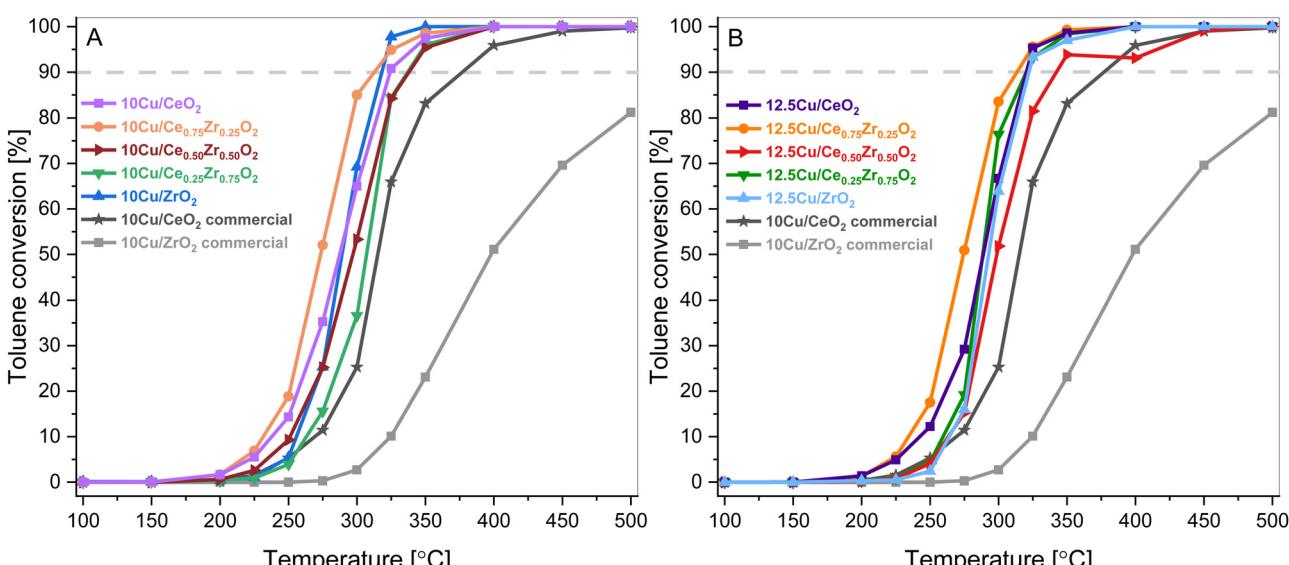


Fig. 9 Toluene conversions as a function of reaction temperature over the 10 wt% Cu (A) and 12.5 wt% Cu (B) Ce_xZr_{1-x}O₂ materials.



selectivity of all studied materials towards complete combustion of toluene.

As can be seen in Fig. 9, all catalysts outperform the 10 wt% Cu-loaded CeO_2 and ZrO_2 commercial supports that are used as reference catalysts. Consequentially, it is believed that the increased surface area due to the hydrothermal synthesis of the supports is beneficial for the VOC combustion. On top of that, the toluene conversion did not exceed 85% for the commercial CeO_2 material. This inferior result is also coupled with a CO selectivity of around 4% at 300 °C. Furthermore, when analyzing both the 10 wt% and 12.5 wt% Cu series, Cu supported on $\text{Ce}_{0.75}\text{Zr}_{0.25}\text{O}_2$ shows the best results. This is also clearly visible in Fig. 10, which depicts a graphical representation of the temperatures where a toluene conversion of 20% (T_{20}), 50% (T_{50}) and 90% (T_{90}) is achieved over the different catalysts. It is apparent that for both the 10 wt% Cu and 12.5 wt% Cu supported $\text{Ce}_{0.75}\text{Zr}_{0.25}\text{O}_2$ catalysts, very similar results are obtained. More specifically, in the case of 10 wt% Cu/ $\text{Ce}_{0.75}\text{Zr}_{0.25}\text{O}_2$, the T_{20} , T_{50} and T_{90} are observed at 251 °C, 273 °C and 313 °C, respectively, and for 12.5 wt% Cu similar conversions are achieved at 252 °C, 274 °C and 313 °C. This indicates that for these materials the presence of an increased amount of Cu does not improve the catalytic performance and the excess of the introduced active phase does not allow the process to be accelerated. A similar statement can be made when analyzing the CeO_2 , $\text{Ce}_{0.50}\text{Zr}_{0.50}\text{O}_2$ and ZrO_2 based catalysts. With the exception of the T_{90} of CeO_2 , which is 5 °C higher for 10 wt% Cu (257 °C) in comparison to 12.5 wt% Cu (261 °C), all reported values for the T_{20} , T_{50} and T_{90} of the corresponding catalysts are slightly lower or equal. Only in the case of the $\text{Ce}_{0.25}\text{Zr}_{0.75}\text{O}_2$ based catalysts the presence of an increased amount of Cu shows a significant drop in both the T_{50} , which drops over 6% from 307 °C to 288 °C and the T_{90} , which drops around 5% from 337 °C to 320 °C. This can most likely be ascribed to the fact that a higher amount of Cu appears to

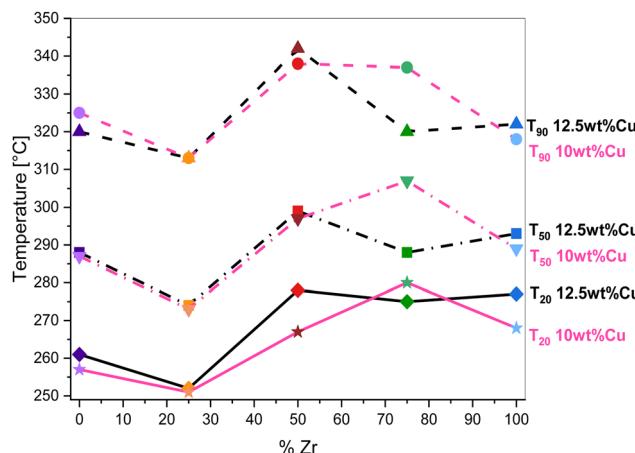


Fig. 11 Stability runs over the 12.5Cu/ $\text{Ce}_{0.75}\text{Zr}_{0.25}\text{O}_2$ catalyst at 300 °C and 325 °C under a dry atmosphere switched after 20 h to a wet atmosphere (~10 vol% H_2O).

stabilize the monoclinic crystal structure, as was observed in the X-ray diffraction measurements.

As shown in the H_2 -TPR results, $\text{Ce}_{0.75}\text{Zr}_{0.25}\text{O}_2$ has the highest reducing capacity among the tested supports. This feature is most likely induced by the presence of Zr^{4+} substituted into the CeO_2 lattice, which thus creates smaller, more easily reducible crystallites. Consequently, it may favor the redox equilibrium at the phase boundary: $\text{Cu}^{2+} + \text{Ce}^{3+} = \text{Cu}^+ + \text{Ce}^{4+}$, which improves the oxygen storage capacity and promotes the surface reaction.

The most active 12.5Cu/ $\text{Ce}_{0.75}\text{Zr}_{0.25}\text{O}_2$ catalyst was tested in stability runs performed at two different temperatures, namely 300 °C and 325 °C. The catalytic measurements were carried out for 20 h in a dry atmosphere, after which 10 vol% of water vapor was additionally introduced into the system. Under the wet conditions, the catalytic test was continued for another 20 h. The collected results are presented in Fig. 11 as the toluene conversion vs. time-on-stream. In the initial period of the reaction a slight improvement in activity is clearly visible. In more detail, at 300 °C the toluene conversion increases from 81.2% to 82.7%, and at 325 °C from 96.3% to 97.2%. The system itself needs around 4 to 5 hours for full stabilization. Most likely, during this induction period the surface is reorganized towards the formation of the previously described redox equilibrium. After introducing a relatively high content of water vapor after 20 h, a decrease in activity of 4.1% (up to 78.6%) at 300 °C and 0.8% (up to 96.4%) at 325 °C is noted. However, despite the presence of a huge excess of H_2O molecules in the reaction system, the activity of the tested catalyst is still very high.

4. Conclusions

This study reports for the first time in the literature the successful deposition of Cu onto $\text{Ce}_x\text{Zr}_{1-x}\text{O}_2$ mixed metal oxides using the water-based ammonia driven deposition precipitation (ADP) method. Five different $\text{Ce}_x\text{Zr}_{1-x}\text{O}_2$ ($x = 0, 0.25, 0.5, 0.75, 1$) mixed metal oxide supports were synthesized using

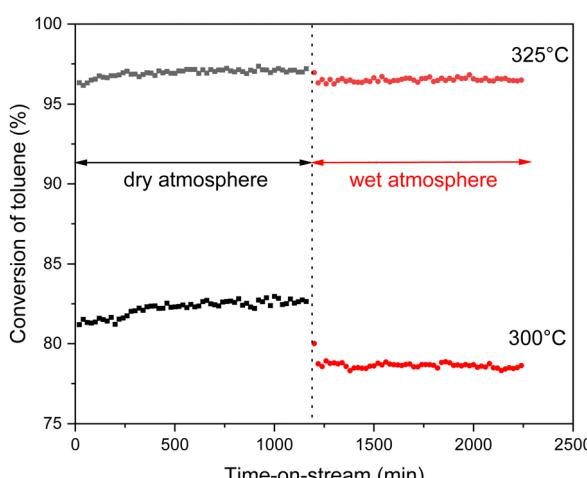


Fig. 10 Overview of temperatures at which a toluene conversion of 20% (T_{20}), 50% (T_{50}) and 90% (T_{90}) is reached over the Cu-supported $\text{Ce}_x\text{Zr}_{1-x}\text{O}_2$ materials.



a soft template assisted hydrothermal approach. The resulting supports were proven to be mesoporous and highly porous, with a much higher surface area and pore volume compared to commercial ceria and zirconia. In order to study the influence of the varying structural and textural properties of the different supports on the ADP Cu deposition, both 10 wt% and 12.5 wt% of Cu was deposited. These loadings were deliberately selected, because a 10 wt% Cu loading is proven to be effective for VOC combustion and 12.5 wt% is the highest attainable Cu loading that ensures a strong metal support interaction with the utilized supports without a significant loss of the active element upon ADP synthesis.

By studying the characteristics of the Cu-based catalysts it was clear that the structure of the support had an impact on the properties of the Cu active phase that was deposited onto the support material, causing differences in the reductive behaviour of the catalysts. However, for all catalysts, the presence of Cu significantly enhanced the redox activity. The strong synergy between Cu and the ceria–zirconia mixed metal oxide supports obtained by using the ADP deposition method resulted in high catalytic conversion of toluene and 100% selectivity towards H₂O and CO₂. On top of that, all catalysts outperformed the commercial CeO₂ and ZrO₂ reference samples onto which 10 wt% of Cu was deposited following the same ADP deposition protocol. Cu supported on Ce_{0.75}Zr_{0.25}O₂ yielded the best results for the catalytic combustion of toluene with a *T*₉₀ of 313 °C, which is identical for both the 10 wt% Cu and 12.5 wt% Cu containing samples. Clearly, the catalytic activity tests highlighted that an increased amount of Cu up to 12.5 wt% typically does not yield an improved catalytic performance. Finally, stability runs at 300 °C and 325 °C in the presence of 10 vol% of water vapor for 20 h indicated that the activity of the 12.5Cu/Ce_{0.75}Zr_{0.25}O₂ catalyst remains very high. Thus proving the potential of a noble metal-free Cu/Ce_{0.75}Zr_{0.25}O₂ catalyst, synthesized using a water-based synthesis protocol that involves the ADP deposition of Cu, for VOC combustion.

Author contributions

W. Van Hoey: conceptualization, methodology, investigation, formal analysis, validation, writing – original draft, writing – review & editing. A. Rokicińska: methodology, investigation, formal analysis, validation, writing – review & editing. M. Dębosz: methodology, investigation, formal analysis, validation, writing – review & editing. I. Majewska: investigation, writing – review & editing. P. Kuśtrowski: conceptualization, funding acquisition, investigation, methodology, writing – original draft, writing – review & editing, supervision. P. Cool: conceptualization, funding acquisition, investigation, methodology, writing – review & editing, supervision.

Data availability

Data will be made available upon request to the corresponding author.

Conflicts of interest

There are no conflicts to declare.

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Notes and references

- 1 A. Rokicińska, M. Drozdek, B. Dudek, B. Gil, P. Michorczyk, D. Brouri, S. Dzwigaj and P. Kuśtrowski, *Appl. Catal., B*, 2017, **212**, 59–67.
- 2 Rahul, A. K. Mathur and C. Balomajumder, *Bioresour. Technol.*, 2013, **142**, 9–17.
- 3 T. Kondratowicz, M. Drozdek, M. Michalik, W. Gac, M. Gajewska and P. Kuśtrowski, *Appl. Surf. Sci.*, 2020, **513**, 145788.
- 4 P. Kuśtrowski, A. Rokicińska and T. Kondratowicz, *Adv. Inorg. Chem.*, 2018, **72**, 385–419.
- 5 T. Kondratowicz, M. Drozdek, A. Rokicińska, P. Natkański, M. Michalik and P. Kuśtrowski, *Microporous Mesoporous Mater.*, 2019, **279**, 446–455.
- 6 H. Wang, W. Yang, P. Tian, J. Zhou, R. Tang and S. Wu, *Appl. Catal., A*, 2017, **529**, 60–67.
- 7 X. Hu, Z. Zhang, Y. Zhang, L. Sun, H. Tian and X. Yang, *Eur. J. Inorg. Chem.*, 2019, 2933–2939.
- 8 B. Zhou, Q. Ke, M. Wen, T. Ying, G. Cui, Y. Zhou, Z. Gu and H. Lu, *J. Rare Earths*, 2023, **41**(8), 1171–1178.
- 9 B. Zhao, Y. Jian, Z. Jiang, R. Albilali and C. He, *Chin. J. Catal.*, 2019, **40**, 543–552.
- 10 A. Aguirre, E. L. Fornero, A. Villarreal and S. E. Collins, *Catal. Today*, 2022, **394–396**, 225–234.
- 11 H. Wang, Y. Zhang, M. Wu, H. Xu, X. Jin, J. Zhou and Z. Hou, *Ind. Eng. Chem. Res.*, 2020, **59**, 20316–20324.
- 12 T. Liu, H. Yan, J. Xu, X. Xu, Y. Lv, X. Fang and X. Wang, *Catal. Today*, 2023, **421**, 114177.
- 13 K. Xie, Z. Wang, B. Jiang, B. Zhao and S. Zuo, *Appl. Catal., A*, 2022, **639**, 118641.
- 14 Y. Dong, J. Zhao, J. Y. Zhang, Y. Chen, X. Yang, W. Song, L. Wei and W. Li, *Chem. Eng. J.*, 2020, **388**, 124244.
- 15 J. Yun, L. Wu, Q. Hao, Z. Teng, X. Gao, B. Dou and F. Bin, *J. Environ. Chem. Eng.*, 2022, **10**, 107847.



16 Y. Zhang, Z. Zeng, Y. Li, Y. Hou, J. Hu and Z. Huang, *Fuel*, 2021, **288**, 119700.

17 M. Ma, X. Feng, R. Yang, L. Li, Z. Jiang, C. Chen and C. He, *Fuel*, 2022, **317**, 123574.

18 Q. Wang, K. L. Yeung and M. A. Bañares, *Catal. Today*, 2020, **356**, 141–154.

19 M. Ozawa, M. Misaki, M. Iwakawa, M. Hattori, K. Kobayashi, K. Higuchi and S. Arai, *Catal. Today*, 2019, **332**, 251–258.

20 Y. Liu, J. Yang, J. Yang, L. Wang, Y. Wang, W. Zhan, Y. Guo, Y. Zhao and Y. Guo, *Appl. Surf. Sci.*, 2021, **556**, 149766.

21 K. J. Kim, Y. L. Lee, H. S. Na, S. Y. Ahn, J. O. Shim, B. H. Jeon and H. S. Roh, *Catalysts*, 2020, **10**, 6–9.

22 S. Y. Ahn, H. S. Na, K. W. Jeon, Y. L. Lee, K. J. Kim, J. O. Shim and H. S. Roh, *Catal. Today*, 2020, **352**, 166–174.

23 S. Adak, J. Rabeah, R. Ranjan, T. S. Khan, M. K. Poddar, R. K. Gupta, T. Sasaki, S. Kumar, A. Bordoloi, C. S. Gopinath, A. Brückner and R. Bal, *Appl. Catal., A*, 2021, **624**, 118305.

24 C. A. Chagas and M. Schmal, *Int. J. Hydrogen Energy*, 2022, **47**, 8858–8866.

25 H. S. Kim, H. J. Kim, J. H. Kim, J. H. Kim, S. H. Kang, J. H. Ryu, N. K. Park, D. S. Yun and J. W. Bae, *Catalysts*, 2022, **12**(1), 63.

26 I. Atribak, A. Bueno-López and A. García-García, *J. Catal.*, 2008, **259**, 123–132.

27 J. I. Gutiérrez-Ortíz, B. de Rivas, R. López-Fonseca and J. R. González-Velasco, *Appl. Catal., B*, 2006, **65**, 191–200.

28 E. Mamontov, T. Egami, R. Brezny, M. Koranne and S. Tyagi, *J. Phys. Chem. B*, 2000, **104**, 11110–11116.

29 Q. F. Deng, T. Z. Ren, B. Agula, Y. Liu and Z. Y. Yuan, *J. Ind. Eng. Chem.*, 2014, **20**, 3303–3312.

30 B. Dou, D. Liu, Q. Zhang, R. Zhao, Q. Hao, F. Bin and J. Cao, *Catal. Commun.*, 2017, **92**, 15–18.

31 Z. Hou, J. Feng, T. Lin, H. Zhang, X. Zhou and Y. Chen, *Appl. Surf. Sci.*, 2018, **434**, 82–90.

32 C. Zhou, H. Zhang, Z. Zhang and L. Li, *Appl. Surf. Sci.*, 2021, **539**, 148188.

33 Y. Shen, J. Deng, X. Hu, X. Chen, H. Yang, D. Cheng and D. Zhang, *Environ. Sci. Technol.*, 2023, **57**, 1797–1806.

34 B. Dou, R. Zhao, N. Yan, C. Zhao, Q. Hao, K. S. Hui and K. N. Hui, *Mater. Chem. Phys.*, 2019, **237**, 121852.

35 Z. Zhang, G. Hu, C. Zhao, X. Wei, B. Dou, W. Liang and F. Bin, *Fuel*, 2023, **341**, 127760.

36 C. Zhang, C. Zhao, R. Kang, Q. Hao, B. Dou and F. Bin, *Carbon Resour. Convers.*, 2023, **6**(4), 255–261.

37 P. S. Barbato, S. Colussi, A. Di Benedetto, G. Landi, L. Lisi, J. Llorca and A. Trovarelli, *J. Phys. Chem. C*, 2016, **120**, 13039–13048.

38 B. Song, C. Li, X. Du, S. Li, Y. Zhang, Y. Lyu and Q. Zhou, *Fuel*, 2021, **306**, 121654.

39 E. Moretti, M. Lenarda, P. Riello, L. Storaro, A. Talon, R. Frattini, A. Reyes-Carmona, A. Jiménez-López and E. Rodríguez-Castellón, *Appl. Catal., B*, 2013, **129**, 556–565.

40 X. Guo, A. Yin, W. L. Dai and K. Fan, *Catal. Lett.*, 2009, **132**, 22–27.

41 Q. Xin, A. Papavasiliou, N. Boukos, A. Glisenti, J. P. H. Li, Y. Yang, C. J. Philippopoulos, E. Poulakis, F. K. Katsaros, V. Meynen and P. Cool, *Appl. Catal., B*, 2018, **223**, 103–115.

42 A. Papavasiliou, T. Van Everbroeck, C. Blonda, B. Oliani, E. Sakellis, P. Cool, P. Canu and F. K. Katsaros, *Fuel*, 2022, **311**, 122491.

43 T. Van Everbroeck, A. Papavasiliou, R. G. Ciocarlan, E. Poulakis, C. J. Philippopoulos, E. O. Jardim, J. Silvestre-Albero, E. Sakellis, P. Cool and F. K. Katsaros, *Catalysts*, 2022, **12**(6), 628.

44 B. Dou, D. Yang, T. Kang, Y. Xu, Q. Hao, F. Bin and X. Xu, *Carbon Resour. Convers.*, 2021, **4**, 55–60.

45 W. Yu, Q. Zhou, H. Wang, Y. Liu, W. Chu, R. Cai and W. Yang, *J. Mater. Sci.*, 2020, **55**, 2321–2332.

46 O. Mangla and S. Roy, *Proceedings*, 2019, **3**(1), 10.

47 S. N. Basahel, M. Mokhtar, E. H. Alsharaeh, T. T. Ali, H. A. Mahmoud and K. Narasimharao, *Catalysts*, 2016, **6**(4), 57.

48 J. L. Ayastuy, A. Gurbani, M. P. González-Marcos and M. A. Gutiérrez-Ortíz, *Int. J. Hydrogen Energy*, 2010, **35**, 1232–1244.

49 J. F. Xu, W. Ji, Z. X. Shen, W. S. Li, S. H. Tang, X. R. Ye, D. Z. Jia and X. Q. Xin, *J. Raman Spectrosc.*, 1999, **30**, 413–415.

50 B. M. Reddy, P. Bharali, P. Saikia, A. Khan, S. Loridan, M. Muhler and W. Grünert, *J. Phys. Chem. C*, 2007, **111**, 1878–1881.

51 F. Goubin, X. Rocquefelte, M. Whangbo, Y. Montardi, R. Brec and S. Jobic, *Chem. Mater.*, 2004, **16**, 662–669.

52 J. Du, W. Chen, G. Wu, Y. Song, X. Dong, G. Li, J. Fang, W. Wei and Y. Sun, *Catalysts*, 2020, **10**, 1–11.

53 A. G. Kong, H. W. Wang, X. Yang, Y. W. Hou and Y. K. Shan, *Microporous Mesoporous Mater.*, 2009, **118**, 348–353.

54 K. V. R. Chary, G. V. Sagar, C. S. Srikanth and V. V. Rao, *J. Phys. Chem. B*, 2007, **111**, 543–550.

55 R. Si, Y. W. Zhang, S. J. Li, B. X. Lin and C. H. Yan, *J. Phys. Chem. B*, 2004, **108**, 12481–12488.

56 T. Tsoncheva, R. Ivanova, J. Henych, M. Dimitrov, M. Kormunda, D. Kovacheva, N. Scotti, V. D. Santo and V. Štengl, *Appl. Catal., A*, 2015, **502**, 418–432.

57 G. Ranga Rao and H. Ranjan Sahu, *Proc. – Indian Acad. Sci., Chem. Sci.*, 2001, **113**, 651–658.

58 X. Yang, X. Ma, X. Yu and M. Ge, *Appl. Catal., B*, 2020, **263**, 118355.

59 M. Simonov, Y. Bespalko, E. Smal, K. Valeev, V. Fedorova, T. Krieger and V. Sadykov, *Nanomaterials*, 2020, **10**, 1–19.

60 J. L. Cao, Y. Wang, T. Y. Zhang, S. H. Wu and Z. Y. Yuan, *Appl. Catal., B*, 2008, **78**, 120–128.

61 Y. Shao, T. Wang, K. Sun, Z. Zhang, L. Zhang, Q. Li, S. Zhang, G. Hu and X. Hu, *Green Energy Environ.*, 2021, **6**, 557–566.

62 L. Liu, Z. Yao, B. Liu and L. Dong, *J. Catal.*, 2010, **275**, 45–60.

63 P. Kaminski, M. Ziolek and J. A. Van Bokhoven, *RSC Adv.*, 2017, **7**, 7801–7819.

64 M. M. Natile and A. Glisenti, *Surf. Sci. Spectra*, 2006, **13**, 17–30.

65 A. Rokicińska, P. Majerska, M. Drozdek, S. Jarczewski, L. Valentin, J. Chen, A. Slabon, S. Dzwigaj and P. Kuśtrowski, *Appl. Surf. Sci.*, 2021, **546**, 149148.

