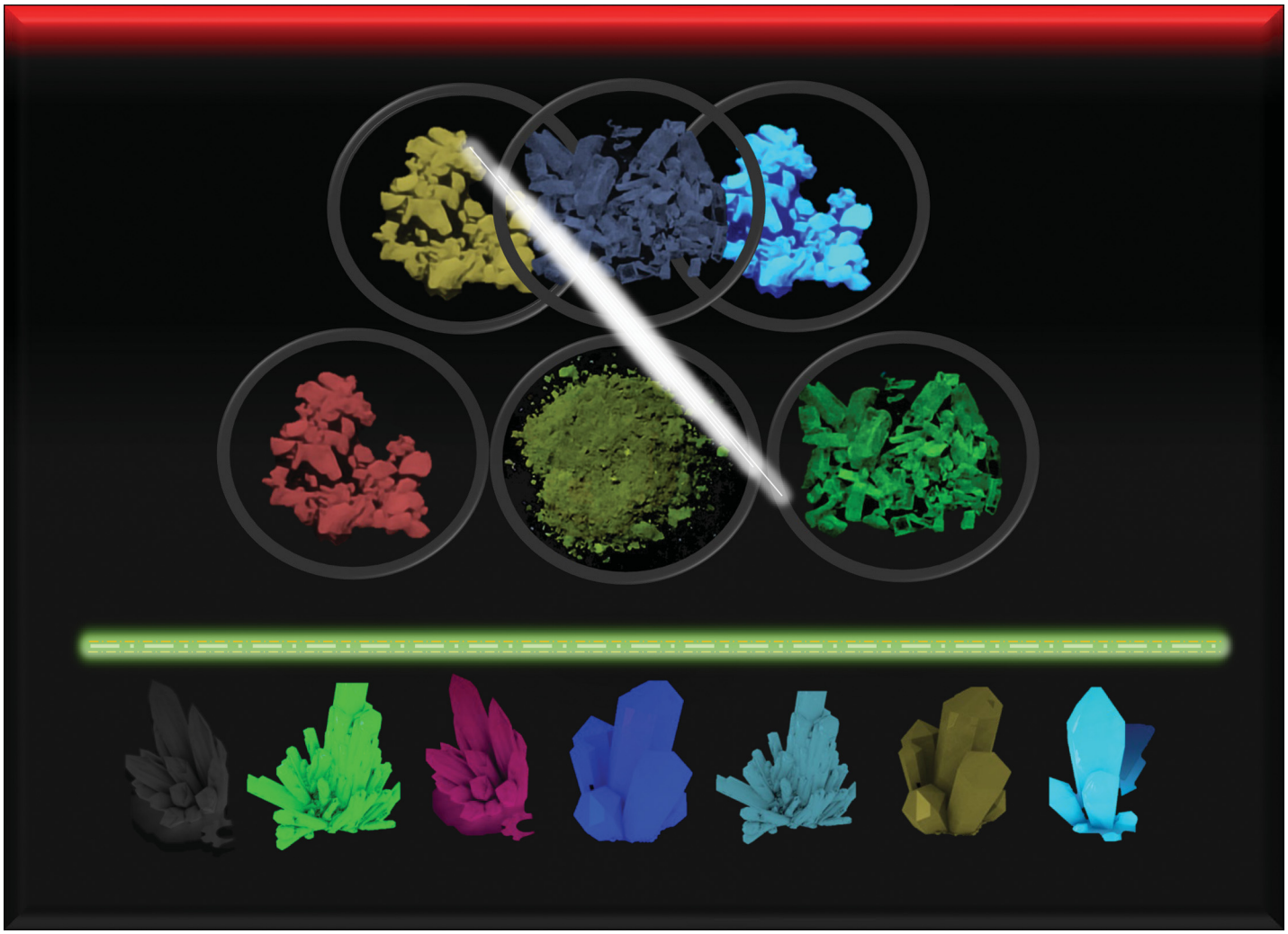

Showcasing research from Professor Aijaz A. Dar's Crystal Engineering laboratory, Department of Chemistry, University of Kashmir, Jammu & Kashmir, India.

Engineering the solid-state luminescence of organic crystals and cocrystals

Crystal engineering has been used to investigate the combined impact of functionalization and co-crystallization on the photophysical properties of organic solids. Structure analyses attribute the emission tuning to the different aggregation patterns establishing structure–property relationships.

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Engineering the solid-state luminescence of organic crystals and cocrystals†

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Fine-tuning the solid-state emission of organic materials is a topic of immense commercial significance and academic interest. Of the various approaches employed to achieve emission-tuning, the co-crystal approach is less reported. In this study, we integrate the effect of functionalization and co-crystallization to investigate the optical properties. Three functionalized pyridyl-hydrazone molecules **1a** (NO_2), **2a** (CN) and **3a** (Br) have been synthesized and further utilized for cocrystal development with 5-sulfosalicylic acid (5-SSA-2H) to obtain organic salts **1–3**, respectively. **1a–3a** exhibit varied aggregation-induced emission (AIE) behavior, which is further tuned through co-crystallization. Emission quenching in **1a** is attributed to long-range π–π stacking between the organic molecules while no π–π stacking interactions are observed in **2a** (464 nm) and **3a** (442 nm), resulting in their emissive behavior. Co-crystallization leads to more regulation of the emission wavelengths as **1a** undergoes emission turn-on in the organic salt **1** (467 nm), and exhibits brownish luminescence; meanwhile, solid-state emission of **2a** and **3a** is moderately and significantly red-shifted in 2 (472 nm) and 3 (484 nm), respectively. The emission turn-on in **1** and red-shift in **2** is attributed to J-aggregate formation in their solid state and a significant red-shift of **3** is attributed to the formation of isolated head-to-tail dimers or excimers in the crystal lattice. The results are further supported with powder-X-ray diffraction, AIE and Hirshfeld studies.

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Introduction

Organic solid-state emitters have been reported since the 1896 observations of Schmidt; however, this area has garnered attention since the concept of aggregation-induced emission (AIE) was floated by Tang.[1,2] This phenomenon expands the application range of these materials as organic light-emitting diodes (OLEDs),[3–5] organic field effect transistors (OLETs),[6–9] organic photovoltaics (OPVs),[10] anti-counterfeiting agents[11] and fluorescent sensors.[12–14] Organic AIE materials are cost-effective and provide the advantage of molecular design to achieve easy property modulations, good solution processability and facile post-usage disposal over their inorganic counterparts.[15] The design and synthesis of novel organic luminescent crystalline materials can be realized through crystal engineering, which represents a greener and more economical alternative to organic synthesis. The resultant crystalline materials provide an opportunity of structure analyses and establishing of the structure-optical property relationship,[16] and the photophysical parameters (emission wavelength, quantum yields and life-time) of these materials are a function of intermolecular contacts (hydrogen bonds, charge transfer, π–π interactions, *etc.*) and the nature of aggregation (H, J, X, I-aggregates).[17]

Co-crystals are an important outcome of crystal engineering, and represent an emergent class of materials with a broad spectrum of applications and prospective opportunities as pharmaceuticals,[18–20] devices,[21–23] multiple-stimuli responsive materials,[24–26] semiconductors,[27] lasers,[28] optoelectronics,[29] *etc.* Emission tuning in single component crystals can be achieved through functionalisation and polymorphism, both of these approaches are well reported. While functionalization involves electronic effects, polymorphism involves the packing effect on the optical properties.[29–33] The co-crystal approach is a more facile process through which multi-component crystals can be prepared with better control over design and properties. Charge transfer co-crystals, based on π–π interactions, have been reported and investigated for their electronic and optical properties. Emission tuning in such cocrystals has been reported through coformer selection. As one of the first reports, Jones and co-workers have demonstrated that emission tuning in 1,4-bis-*p*-cyanostyrylbenzene was realised through co-crystallization with a

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† Electronic supplementary information (ESI) available: FT-IR, powder X-ray diffraction, Hirshfeld studies, structure diagrams, absorption and emission spectra, AIE and DLS studies and crystallographic table. CCDC 2255995–2255997. For ESI and crystallographic data in CIF or other electronic format see DOI: https://doi.org/10.1039/d3ma00853c

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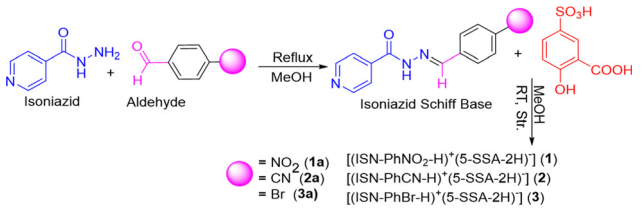

Scheme 1 Schematic representation of the reaction pathway.

range of suitable coformers.[34] Since then, numerous charge transfer complexes have been developed and reported to exhibit emission tuning.[35–42] Chopra and co-workers have demonstrated emission tuning in organic cocrystals utilizing the understanding of intermolecular interactions.[43] Oxborrow's group has also utilized π-stacked interactions to develop MASER cocrystals.[44] Similarly, arene-perfluorene interactions involving a perfluoro aromatic acceptor and a polyaromatic hydrocarbon do not undergo charge transfer interactions and have been reported to achieve blue-shifted emissions, as compared to red-shifted emissions in charge transfer complexes.[45,46] Limited choice of acceptor molecules, however limits the scope of cocrystals developed based on π interactions.

Hydrogen bonding is another significant intermolecular interaction widely used for cocrystal and molecular salt formation, and hydrogen bonded cocrystals/salts with emission or emission tuning, however, are lesser reported. Draper *et al.* demonstrated the emission switching through hydrogen bonding tendencies of solvent molecules.[47] Similarly, Yang and Gazit have demonstrated the use of hydrogen bonds to tune the emission.[48] These systems, however, have been reported randomly and there are no established strategies to design emission in hydrogen bonded complexes. Recently we have reported emission tuning of the organic precursor by its cocrystallization, as the pyridyls of the same shape but the different positions of the nitrogen lead to different packing arrangements.[49,50] However, advancement in the area remains limited and requires more vigorous endeavours.

In this study, we have attempted to determine the impact of substitution and cocrystallization on the AIE of new solid forms. Planar and linear electronegative substituents were attached to the hydrazone Schiff base of isoniazid to obtain **1a–3a**, which were further modified by cocrystallization with 5-sulfosalicylic acid (5-SSA-2H) to synthesize **1–3**, Scheme 1. Optical studies of **1a–3a** and **1–3** have been performed in detail and the solid-state emission tunning in these materials due to substituent and packing effects has been explained with the help of diffraction and Hirshfeld studies. Furthermore, the AIE studies of both **1a–3a** and **1–3** have been reported.

Experimental section

Methods and materials

Isoniazid (99%, Sigma Aldrich), 4-nitrobenzaldehyde (99%, Sigma Aldrich), 4-formylbenzonitrite (99%, Sigma Aldrich), p-bromobenzaldehyde (97%, Sigma Aldrich) and 5-sulphosalicylic acid (5-SSA-3H) (99%, Sigma Aldrich) were purchased from Sigma Aldrich and used without further purification. The Schiff bases **1a–3a** have been synthesized by refluxing the methanolic solution of isoniazid and aldehydes for 2–3 hours. Methanol (SD fine), DMF (SRL), and distilled water have been used as solvents for crystallization and/or for photo-physical studies. Melting points have been determined on the MP70 melting point system capillary apparatus (Mettler Toledo) in closed-end capillaries. Infrared spectroscopic data for molecular salts has been obtained using a 630 FT-IR (4000–650 cm^{-1}) in ATR mode. Ground crystals of the products are placed on the crystal plate of the infrared instrument, to record the spectrum. Diffuse reflectance measurements of the products have been recorded on a Shimadzu-2600 spectrometer on $BaSO_4$ discs and converted to Kubelka–Munk function and Tauc plots in origin using an appropriate conversion formula.[51] The absorption studies have been carried out on a Shimadzu-2600 Spectrophotometer.

SC-XRD studies

Single-crystal data were collected on a Rigaku Saturn CCD diffractometer using a graphite monochromator (Mo Kα, λ = 0.71073 Å). The selected crystals were mounted on the tip of a glass pin using mineral oil and placed in the cold flow produced with a Cryo-cooling device. Complete hemispheres of data were collected using ω and φ scans (0.3°, 16 s perframe). Integrated intensities were obtained with Rigaku Crystal Clear-SM Expert 2.1 software, and they were corrected for absorption correction. Structure solution and refinement were performed with the SHELX package. The structures were solved by direct methods and completed by iterative cycles of ΔF syntheses and full-matrix least-squares refinement against F. Crystal refinement parameters of **1a–3a** are given in Table S1, ESI.†

Fluorescence measurements

Solid state emission was recorded on a Shimadzu RF-5301PC Spectrofluorometer by making use of $BaSO_4$ discs on which compounds **1a–3a**, **1–3,** and 5-SSA-2H were placed and emission was recorded at room temperature in normal mode by exciting them at 320 nm. Emission in the solution state was recorded on the same instrument by taking 10^{-3} M solution in the cuvettes and each was excited at 300 nm.

Powder X-ray studies

Powder (PXRD) data of compounds **1–3** was recorded with a BRUKER-AXS-D8-ADVANCE diffractometer (Cu kα, λ = 1.5406) at room temperature with theta ranging from 10 to 40.

Hirshfeld studies

The Hirshfeld analysis was carried out through CrystalExplorer 17.5 software. Color coding mapped on the d_{norm} surface represents the contacts that indicate short (red color), intermediate (white color), and long contacts (blue color) compared with the sum of van der Waals interactions. The details are provided in the ESI.† file

Synthesis of 1. **1a** was prepared by dissolving 137 mg (1 mmol) of Isoniazid and 151 mg (1 mmol) of p-nitro

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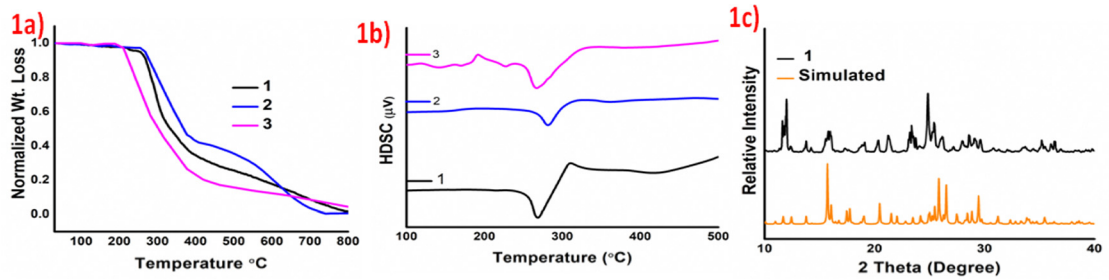

Fig. 1 (a) TGA, (b) DSC of cocrystals **1–3**, and (c) PXRD of cocrystal-**1** with its simulated data.

benzaldehyde and was refluxed for 4 h at 60 °C. The co-crystal **1** was obtained by dissolving **1a** (135 mg, 0.5 mmol) and 5-SSA-2H (127 mg, 0.5 mmol) separately in 10 mL of methanol and the resultant solutions were mixed. The resultant solution was further concentrated by heating for about 10 minutes, and filtered. The filtrate was allowed to slowly evaporate to yield light yellow block-shaped crystals within 3 days. Yield: 55%. M.P. >230 °C, IR, ν, cm^{-1}: 3646 (s) 2872 (s), 1636 (s), 1565 (s) 1028 (s). Abs. peaks 313 (nm) and 400 (nm)

Synthesis of 2. **2a** was prepared by dissolving 137 mg (1 mmol) of isoniazid and 131 mg (1 mmol) of 4-formylbenzonitrile and was refluxed for 4 h at 60 °C. Cocrystal 2 was prepared by dissolving **2a** (127 mg, 0.5 mmol) and 5-SSA-2H (127 mg, 0.5 mmol) separately in 10 mL of methanol and then the solutions were mixed. The resultant solution was further concentrated by heating for about 10 minutes, and filtered. Yellow rectangular/prism-shaped crystals were obtained after 2 days by slow evaporation. Yield: 60%. MP: 281 °C, IR, ν, cm^{-1}: 3653 (s), 2887 (s), 1629 (s), 1203 (s) 1050 (s), 2221 (s). Abs. peak 314 (nm) and 376 (nm)

Synthesis of 3. **3a** was prepared by dissolving 137 mg (1 mmol) of isoniazid and 185 (1 mmol) mg of 4-bromobenzaldehyde and refluxing for 4 h at 60 °C. Cocrystal **3** was prepared by dissolving **3a** (152 mg, 0.5 mmol) and 5-SSA-2H (127 mg, 0.5 mmol) separately in 10 mL of methanol and the solutions were mixed. The resultant solution was further concentrated by heating for about 10 minutes, and filtered. Yellow block-shaped crystals were obtained in 3 days by slow evaporation of the solution. Yield: 60%. MP: 266 °C, IR, ν, cm^{-1} 3708 (s) 2976 (s), 1663 (s), 1249 (s) 1061 (s), 672 (s). Abs peak 314 (nm) and 376 (nm).

Results

To investigate the combined effect of the substituents and crystal packing on the properties, three hydrazone Schiff bases of isoniazid with nitro (**1a**), cyano (**1b**) and bromo (**1c**) substituents at the para-position of the aldehydic end were synthesized and utilized for the preparation of organic salts **1–3**. The formation of the products is validated by Fourier-transform Infrared (FT-IR) spectroscopy. The characteristic absorptions in the spectrum of **1–3** at 1670–1680 cm^{-1} and 1640–1575 cm^{-1} correspond to the stretching vibrations of C=O and C=N, respectively. The absorption bands at 1320–1170 cm^{-1} and 1150–1030 cm^{-1} correspond to the asymmetrical and symmetrical stretching of the sulfonic group, respectively, and the stretching frequency around 2872, 2887 and 2976 cm^{-1} in **1**, **2** and **3**, respectively, corresponds to the protonation of the pyridinium end of the Schiff bases (N^+–H bond), indicating proton transfer or salt formation. Furthermore, the diagnostic peaks for different functional groups are observed at 1576 cm^{-1} and 1391 cm^{-1} for –NO_2, 2219 cm^{-1} for –CN and 696 cm^{-1} for –Br in **1–3**, respectively (Fig. S1–S3, ESI†).

Thermal studies indicate significant stability of the products and the salts **1–3** form melts at temperatures greater than 230 °C. The Thermogravimetric analyses (TGA) and differential scanning analyses (DSA) of the products validate augmented thermal stability as they start to decompose after 230 °C and show endothermic dips in the DTA curves at 267, 281 and 266 °C for **1–3**, respectively, which can be attributed to their melting points. Augmented thermal stabilities of these organic salts may be attributed to the ionic interactions between crystal components, due to proton transfer. The TGA and DSC curves of **1–3** are provided in Fig. 1a and b, and the early irregularity in the DSC curve of **1c** plausibly arises due to the scattering of the product on heating. Powder X-ray diffraction studies of **1–3** have been carried out to understand the phase changes on the size reduction of the crystals. A comparison of the experimental and the simulated data confirms that there is no significant phase change on size reduction from single crystals to microcrystalline phase. Moreover, the appearance of sharp peaks in the powder X-ray diffraction indicates retention of the polycrystalline nature even on size reduction. Experimental and simulated powder X-ray data of product **1** are given in Fig. 1c and for **2** and **3** are provided in Fig. S4 and S5, ESI.†

Optical properties

Solid-state luminescence of materials is an intriguing phenomenon with significant prospective consequences. Besides commercial implications, the process is of great academic interest. The organic salts **1–3**, obtained through cocrystallization of **1a–3a** with 5-SSA-2H, exhibit intriguing colour modulation on formation and the chromic changes are also accompanied by changes in the photo-luminescence. The optical images of **1–3** and their precursors **1a–3a**, recorded under visible and long ultra-violet exposure, are provided in Fig. 2. The solid-state modulation of the photo-physical parameters in **1a–3a** is a consequence of the functionalization and the Schiff base **1a**

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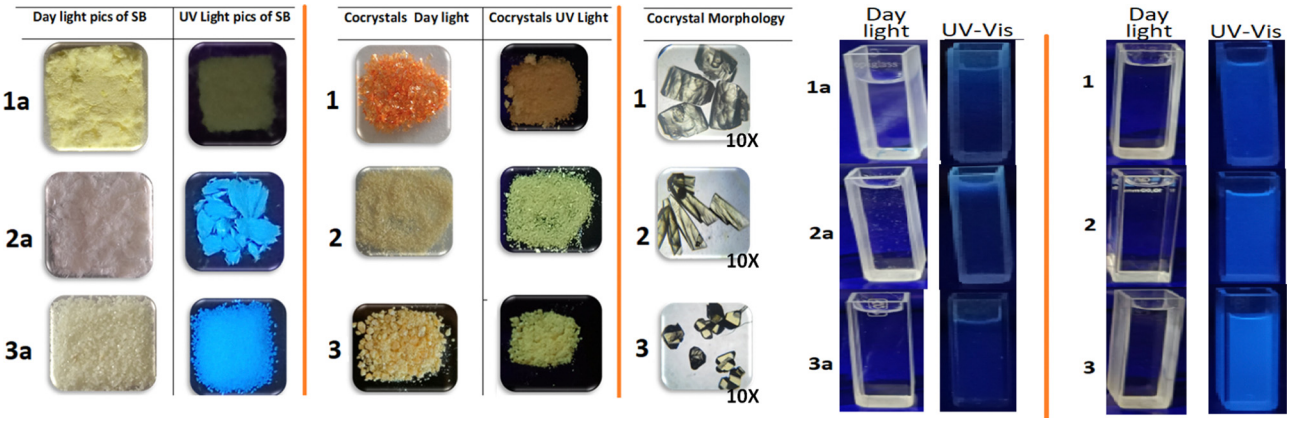

Fig. 2 (Left) Optical images of **1a–3a** and **1–3** recorded under visible and UV exposure and crystal morphologies of **1–3**, at 10× magnification and (Right) optical images of 10^{-3} M aqueous solutions of **1a–3a** and **1–3**, recorded under visible and UV exposure.

with nitro substitution is non-emissive, while **2a** and **3a** exhibit blue and cyan emission, respectively. Co-crystallization of **1a–3a** with blue emitter 5-SSA-2H, further modulates the luminescence of organic salts **1–3**, which undergo red-shift in their emission. The results stipulate the integration of functionalization and crystal packing to achieve fine-tuned AIE in the products.

To understand the role of intra and intermolecular charge transfer interactions, optical studies have been reported in both solution and the solid state. Solution phase studies of **1a–3a** and **1–3**, have been carried out for their 10^{-3} M aqueous solutions, obtained by heating the resultant solutions up to 80 °C until clear solutions were obtained (Fig. 3a). The solid-state studies have been reported by loading the samples on $BaSO_4$ discs. The absorption bands corresponding to the intermolecular charge transfer interactions observed beyond 400 nm are absent in the solution phase spectra of these compounds, implying that the solution phase optical response in the materials arises from the independent molecules, which do not undergo intramolecular charge transfer. Meanwhile the products exhibit broad absorptions beyond 400 nm in the solid state, indicating aggregation-induced intermolecular charge transfer interactions (Fig. 3b). Interestingly, the aqueous solutions of the Schiff base precursors **1a–3a** are weakly emissive, while the corresponding solutions of **1–3** exhibit blue emission with different intensities with the maximum emission centered at 420, 425 and 418 nm for **1–3**, respectively (Fig. 3c). The solid-state luminescence spectra of these compounds exhibit intriguing modulations. The Schiff base **1a** remains non-emissive, while as **2a** (ϕ 0.10%) and **3a** (ϕ 0.07%) emit in the blue region with emission maxima at 464 and 442 nm, respectively. Emission of **1a** on crystallization is switched on in **1** (ϕ 0.14%) and the organic complex exhibits emission with maximum activity at 467 nm. The luminescence of **2** (ϕ 0.13%) and **3** (ϕ 0.12%) is red-shifted *vis-à-vis* their precursors **2a** and **3a** as well as their solution phase luminescence, with λ_{max} values of 472 and 484 nm, respectively. The un-normalized solid-state emission spectra of **1a–3a** and **1–3** are provided in Fig. 3d, and their optical images are given in Fig. 2. The KM plots, *Tauc* plots, and solid-state emission of **1a–3a**, and **1–3** are provided in Fig. S6–S18, ESI.† Comparative solution and solid state emission parameters of the products are provided in Table S2, ESI.†

Solution phase AIE studies, involving incremental addition of a bad solvent to the non-emissive solutions, have been carried out for **1a–3a** and **1–3**, to gain further insights into the emission behavior. The AIE studies have been carried out by dissolution of **1a–3a** and **1–3** in DMF to prepare 10^{-3} M solutions, to obtain non-emissive solutions. Water was further added as a bad solvent to make them emissive; in this case, **1a–3a** do not exhibit any significant emission turn-on at slit-width values from **3** to **10**, while **1–3** exhibit gradual increases in the emission intensity up to a f_w (mole fraction of water) value of 80% and a sharp augmentation in intensity at f_w 90%. The λ_{max} values for AIEgen solutions of **1–3** are nearly identical and located at 400 nm; however, the behavior is different *vis-à-vis* aggregation as **1** undergoes a gradual change in emission turn-on while **2** and **3** undergo abrupt emission turn on with the addition of water. As the AIEgen emission and solid-state emission of **1–3** are not the same, we believe that the AIEgen emission arises from the coformer 5-SSA-2H, which was also confirmed by the AIE studies of the organo-sulfonate. Quantum yields corresponding to the maximum emission values *i.e.*, f_w 90% for AIEgens **1–3** are 72.0%, 45.0%, and 45.2%, respectively. Dynamic light scattering (DLS) studies of these systems validate that the emission turn-on arises due to increase in the particle size. The AIE, DLS plots and optical images under daylight and UV light, of **1–3** are provided in Fig. S19–S25, ESI.†

Solid-state structure analyses

To understand the effect of substituents on the crystal packing and underlying reasons for the variable solid-state emission, single crystal structure analyses of **1–3** have been carried out and compared with previously reported crystal structures of **1a–3a**.[52] The diffraction quality crystals have been obtained directly from the reaction mixture through the slow evaporation method.

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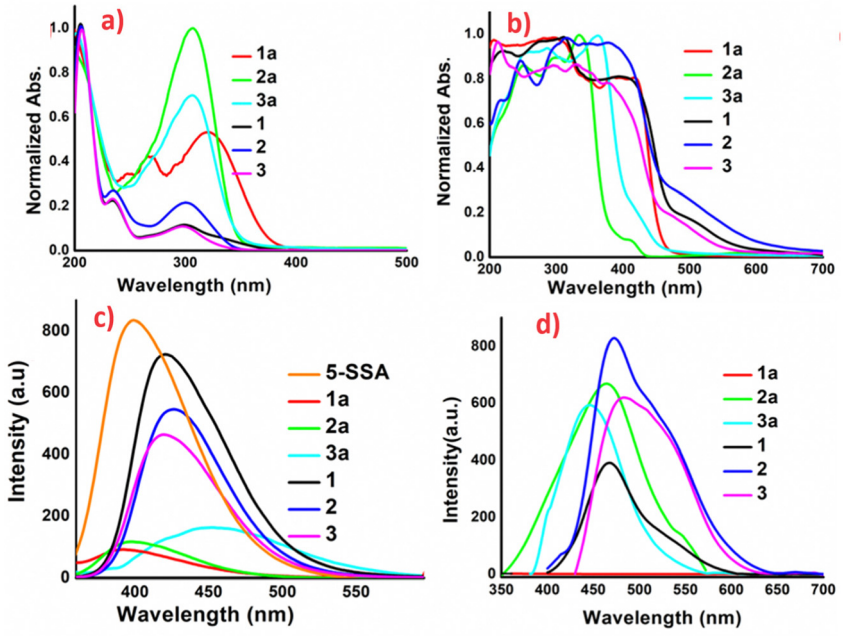

Fig. 3 Photophysical studies: (a) solution phase absorption spectra of 10^{-3} M aqueous solutions, (b) solid phase diffuse-reflectance spectra reported as $BaSO_4$ diluted discs, (c) solution phase emission spectra of 10^{-3} M aqueous solutions and (d) solid-state emission spectra reported as $BaSO_4$ diluted discs, of **1a–3a** and **1–3**.

1 crystallizes in orthorhombic chiral space group $P2_12_12_1$, with two-fold screw symmetry. As anticipated from our previous results,[53–59] the precursors undergo proton transfer to form a robust charge-assisted sulfonate-pyridinium synthon: N1–H1N···O1 [D–H···A: 145.30(1)° D···A: 2.281(1) Å]. Proton transfer is substantiated by nearly equal S–O bond distances in the acid former and relaxed C–N–C bond angle in the base former [122.43(2)°] The isoniazid Schiff base is non-planar and forms convergent trifurcated hydrogen bonds with sulfonate oxygen: O2···H17–C17[150.33°, 2.340 Å], O2···H2N–N2 [133.71(1)°, 2.153(3) Å] and O2···H14–C14 [122.65(2)°, 2.607(1) Å] interactions (Fig. 4a). The free Schiff-base **1a** crystallizes in a nearly planar configuration, and is curved in **1** due to lateral interactions of the NO_2 group with the carbonyl oxygen of the isoniazid residue [O7···N4: 3.028(1) Å]. The nitro group oxygen O8 forms a second hydrogen bond with the pyridinium end of the isoniazid: O8···H1N–N1 [2.929(2) Å], while the carboxylic OH group forms bifurcated interaction with carbonyl oxygen and azo nitrogen: O6–H6A···O7 [166.39(1)°, 1.925(1) Å] and O6–H6A···N3 [117.05(2)°, 2.693(2) Å] to form 2-dimensional hydrogen bonded tapes with cylindrical voids (Fig. S26, ESI†). The tapes further aggregate with weak interactions, including hydrogen bonds: O9···H10–C10 [122.21(1)°, 2.627(2) Å], O5···H6–C6 [157.27(1)°,2.370(1) Å], O3···H20–C20[150.21(2)°, 2.404(1) Å], O5···H6–C6 [157.27(1)°, 2.370(2) Å], O5···H18–C18 [117.63(2)°, 2.636(1) Å], O4···H20–C20 [126.76(1)°, 2.605(1) Å], π–π interactions, and interaction between the sulfonate oxygen O2 and C21 [3.065(1) Å], to form an intriguing 3-dimensional network of hydrogen-bonded tapes (Fig. S27, ESI†).

For 2, the molecular complex crystallizes in a monoclinic $P2_1/n$ symmetry system. The ionic sulfonate-pyridinium synthon is the major intermolecular interaction between the crystal components. The substitution of the nitro functional group in **1** by the nitrile group in the Schiff base in **2**, does not significantly impact the interactions between the co-formers. The trifurcated hydrogen bonding interaction about the sulfonate oxygen and the –C═N–NH– linker of the Schiff base, involving interactions: O2···H14–C14 [130.40(1)°; 2.582(2) Å],

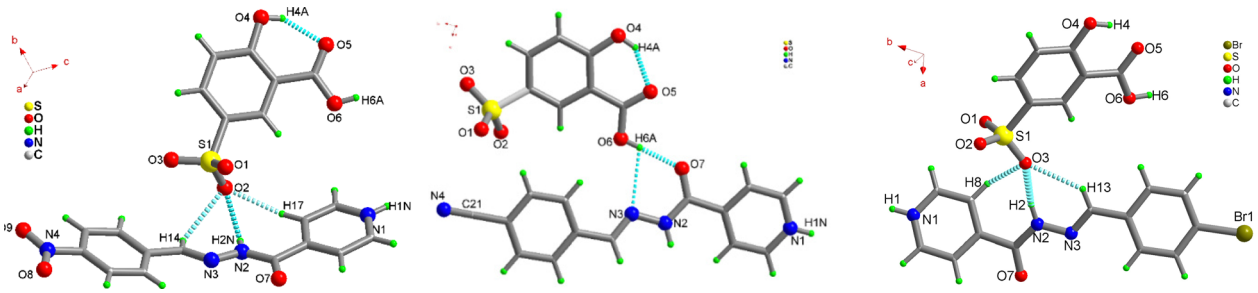

Fig. 4 Molecular structure diagrams of **1**, **2**, and **3**.

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O2···H2N–N2 [144.89(1)°; 2.075(2) Å] and O2···H11–C11 [164.45(1)°; 2.371(1) Å], remains nearly unchanged but unlike in the **1** Schiff base, **2** is nearly planar and not curved (Fig. 4b). The Schiff base **2a** establishes O6–H6A···O7 [174.72(1)°; 2.693(3) Å] and O6–H6A···N3 [108.80(1)°; 3.044(2) Å] contacts on the other side with the hydroxyl of the carboxylate group. The nitrile group does not establish any strong intermolecular contacts.

The organo-sulfonate hydrogen-bonded chains, stabilized by O4–H4A···O3 [116.55(2)°; 2.609(1) Å] hydrogen bonds, with further help from weak intermolecular contacts: C6–H6···O5 [152.25(1)°; 2.414(1) Å], C8–H8···O4 [107.64(1)°; 3.093(2) Å] and C12–H12···N14 [132.58(1)°; 2.447(1) Å], and π–π stacking between Schiff base cations, grow further. The final 3-dimensional hetero-aggregated lattice of **2** is stabilized by a centrosymmetric $R_2^2(10)$ synthon stabilized by C17–H17···N4 [148.82(2)°; 2.604(1) Å] interactions (Fig. S28, ESI†).

3 crystallizes in a monoclinic $P2_1/n$ crystal system with a molecule of each component in the asymmetric unit. The Schiff base is nearly planar and protonated about the pyridyl center and the C–N–C bond angle is relaxed [122.69(2)°] (Fig. 4c). The ionic sulfonate-pyridinium synthon is masked by the Schiff base: O3···H2–N2 [165.74(1)°,1.915(2) Å] and O7···H1–N1 [169.0(2)°,1.874(3) Å]. As observed in **1** and **2**, **3** forms trifurcated hydrogen bonds on one side: O3···H13–C13 [137.40(1)°, 2.661(3) Å] and O3···H8–C8 [129.53(1)°,2.549(3) Å] and bifurcated interactions: O7···H1–N1 [169.0(1)°,1.874(2) Å] and N3···H1–N1 [113.71(3) °, 3.288(2) Å] on the other side of the –C═N–NH– linker. With the aid of a chalcogen bond: O3···Br1 [3.554(1) Å] and O3···N1 [3.135(1) Å], **3** grows into a 3-dimensional solid (Fig. S29, ESI†). The hydrogen-bonded lattice is further stabilized by inverted π-interactions between cationic base formers.

Hirshfeld studies

To further understand the impact of functionalization and cocrystallization on intermolecular interactions and crystal packing Hirshfeld analyses of **1a–3a** and **1–3** have been carried out on Crystal Explorer 17.5 software.[60,61] The Hirshfeld surface generated within a radius of 3.8 Å and calculated over d_{norm} provides a three-dimensional picture of close contacts in a crystal, which can be summarized in their fingerprint plots. Hirshfeld surface properties as well as the fingerprint plots provide significant details of the intermolecular contacts as well as packing of compounds **1a–3a** and **1–3**. Also, the curvedness and shape index provide further information about molecular packing, particularly about π-stacking in the compounds. Fingerprint analyses have been carried out to understand the effect of changing functionality on intermolecular contacts, which are responsible for the crystal packing of the materials. The fingerprint analysis of **1–3** indicates that the main interactions responsible for the formation of these co-crystals are O–H/H–O, H–H and C–H/H–C contributions. The interactions, however, exhibited significant variation with change in the functionality as the O–H/H–O contribution decreases from 41.8% in **1** to 33.4% in **2**, and 29.3% in **3**, which is due to more oxygen centers in **1** and **2** than **3**. The contribution of π–π or C/C is much less in all three solids, but relatively higher in **2** [13.5%] than **1** [8.1%] and **3** [7.5%]. The contributions of other interactions in the precursors **1a–3a** and **1–3** are provided in Fig. 4. The Hirshfeld and fingerprint plots of **1–3** are provided in Fig. S30–S35, ESI.†

Discussion

Aroyl hydrazone was substituted with electron-withdrawing substituents, **1a–3a**, and subsequently co-crystallized with

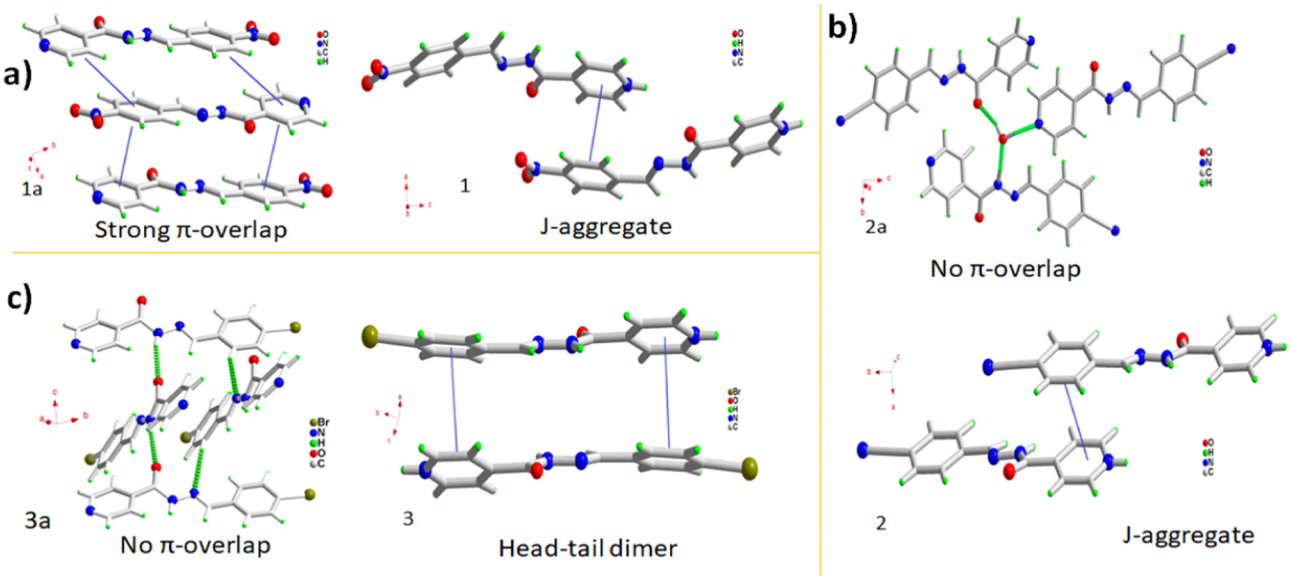

Fig. 5 Packing arrangements of **1–3**. (a) Long range π-overlap in **1a** and J-aggregate formation in **1**, (b) molecular aggregation in **2a** and J-aggregate in **2** and (c) molecular aggregation in **3a** and head-to-tail dimer in **3**.

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5-SSA-2H to understand the combined effect of substitution and cocrystallization on the crystal packing and optical properties. The nitro aroyl hydrazone **1a** undergoes aggregation-caused quenching due to close-packed π–π interactions, while the π-stacking interactions in nitrile **2a** and bromo **3a** aroyl hydrazones are insignificant, leading to aggregation-induced emission (Fig. 5a–c). These Schiff bases undergo remarkable emission tuning on cocrystallization as the resultant organic salts exhibit improved red-shifted luminescence. **1a** undergoes emission turn-on in **1** which in addition to the protonation of its pyridyl functionality can be attributed to the packing modulations on co-crystallization. Instead of complete head-to-tail π-overlap the **1a** molecules in **1** undergo slipped π-interactions forming J-type aggregates (Fig. 5a). The **2a** molecules undergo a moderate red-shift of 8 nm on co-crystallization in **2** which is accompanied by augmented intensity. Though the molecules in **2a** are stacked, they are twisted enough to prevent π-overlap, while the **2a** molecules in **2** aggregate as J-type aggregates with slipped head-to-tail overlap, leading to the red-shifted emission (Fig. 5b). **3a** molecules in **3** undergo a significant red shift of 26 nm. Crystal packing comparison of pure **3a** and its organic complex **3** reveals that the molecules aggregate orthogonally in the pure form, while the **3a** molecules form isolated head-to-tail dimers in the crystal lattice of **3**. The red shift can be plausibly attributed to better π-overlap behavior with the **3a** dimers, and subsequent excimer formation (Fig. 5f).

The solution phase optical studies of the products provide further insights into their optical behavior. The solution phase absorption studies of **1a–3a** and **1–3** do not show any charge transfer interactions indicating their existence as isolated molecular species. The charge transfer interactions emerge in their solid state and the interactions get stronger on cocrystallization. Similarly, the precursors **1a–3a** do not exhibit emission in their aqueous solutions, possibly due to free molecular rotations, while **1–3** exhibit nearly similar emission behavior, which arises due to different contributions from frontier orbitals due to substitution effects. The concentration-dependent solution phase optical studies for the products do not indicate the possibility of excimer emission on aggregation.

The solid-state emission tuning achieved through the combined effect of functionalization and crystallization indicates enough scope for property engineering. Currently, we are working on more variations *vis-a-vis* functional groups and their steric bulk and electronic effects on pyridyl hydrazones with a follow-up of conformer variation. The studies can help to establish the structure–property relationship for the development of desired solid-state emitters for practical applications.

Conclusions

Using the principles of crystal engineering we demonstrate the combined effect of functionalization and crystallization on the structural and optical properties of the organic solids and their co-crystals. The solid-state emission has been engineered and explained *vis-à-vis* crystal structure–propertystructure–property relationship. The emission quenching in **1a** is attributed to strong π–π interactions, which are absent in emissive solids **2a** and **3a**. The emission turn-on of **1a** on co-crystallization in **1** and moderate redshift of **2a** in **2** is attributed to the formation of J-type aggregates in their solid state, while the significant red shift in **3** plausibly arises due to excimer emission of isolated head-to-tail hydrazone dimers. The results represent one of the first attempts to integrate the impact of functional group and co-crystallization on the AIE of organic materials and indicate enough scope for the development of functional solid-state emitters.

Author contributions

D. A. A. has perceived the problem, carried out the crystallographic studies, interpreted the data and prepared the manuscript. S. H. L. has synthesised and characterized the products and generated the data. I. A. and A. A. G. have helped with data plotting/improvisation and manuscript preparation. C. F. has carried out computational studies and helped during the manuscript prepration. A. A. A. has helped during the revision, which involved the reproduction of data and products.

Conflicts of interest

There are no conflicts to declare.

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