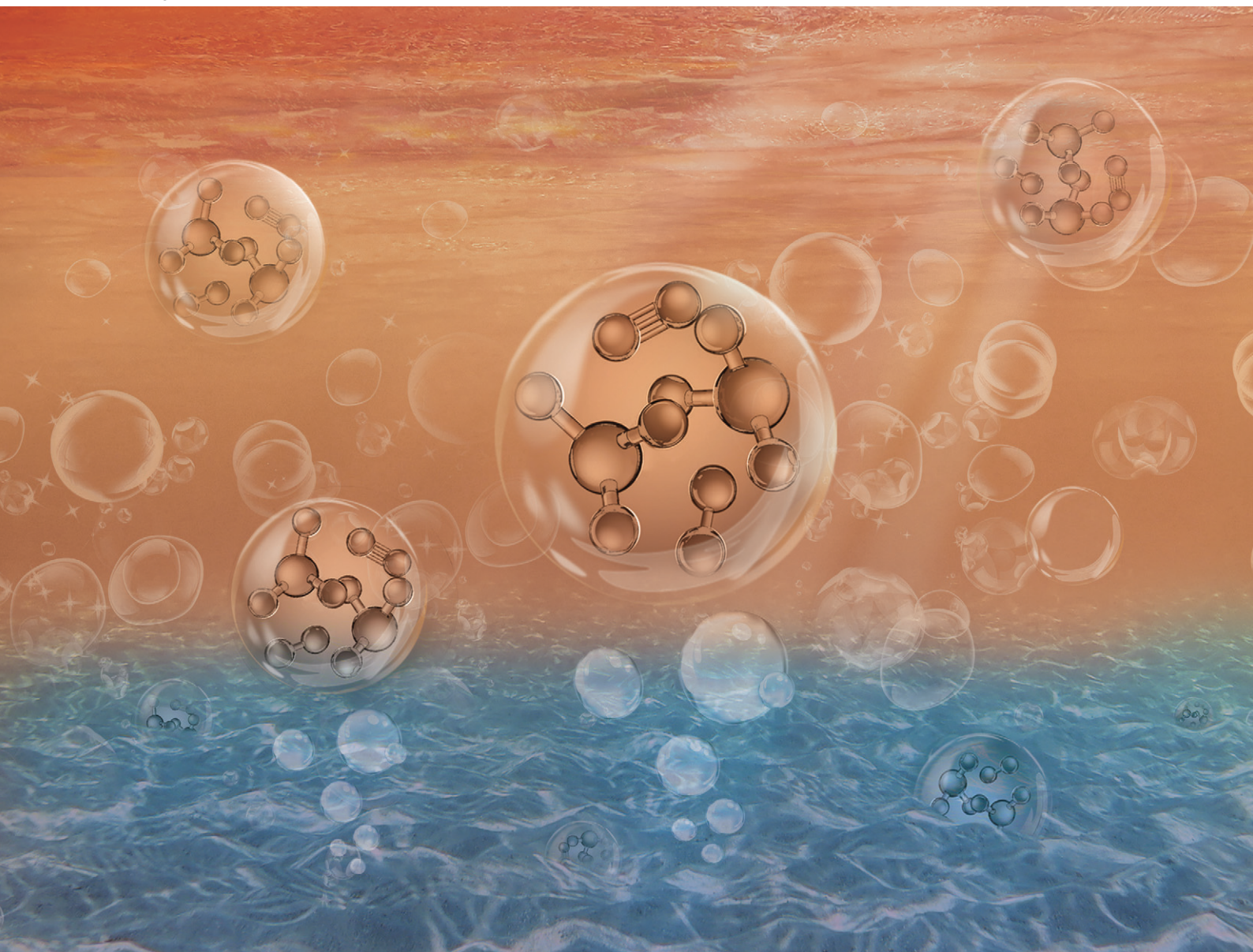


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**PAPER**

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# Molten multi-phase catalytic system comprising Li–Zn alloy and LiCl–KCl salt for nitrogen fixation and ammonia synthesis at ambient pressure†

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Ammonia (NH<sub>3</sub>) is one of the most important synthetic inorganic commodities. The current industrial NH<sub>3</sub> production is dominated by the Haber–Bosch process with high energy cost and CO<sub>2</sub> emission as well as the need for large-scale centralized operation. Liquid metals and molten salts have recently emerged as promising catalytic materials for NH<sub>3</sub> synthesis. Herein, we present a molten system comprising Li–Zn alloy and eutectic LiCl–KCl salt for effective NH<sub>3</sub> synthesis at 400 °C and 1 bar. The 70 mol% Li–Zn liquid alloy activates N<sub>2</sub> dissociation more easily than the pure liquid Zn and the 60 mol% Li–Sn liquid alloy. Effective N<sub>2</sub> fixation by the liquid Li–Zn alloy is followed by the hydrogenation of Li<sub>3</sub>N dissolved in the molten salt above. For the first time, this work reports a volcano-type relationship between the Li<sub>3</sub>N concentration in the molten salt and the NH<sub>3</sub> synthesis rate when feeding H<sub>2</sub> to the molten salt. *Ab initio* molecular dynamics simulations suggest that, within this system, both N<sub>2</sub> cleavage and Li<sub>3</sub>N hydrogenation are quite reactive. Through combined experiments and simulations, this work unravels the molecular mechanisms of nitrogen fixation and ammonia synthesis in the liquid alloy–salt catalytic system, and also demonstrates effective strategies for improving the ammonia synthesis rate. Such a hybrid molten catalytic system offers a promising solution for distributed NH<sub>3</sub> production with low energy cost and CO<sub>2</sub> emission.

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## 1. Introduction

The ammonia (NH<sub>3</sub>) industry plays a significant role in the world economy.<sup>1</sup> More than 80% of synthetic NH<sub>3</sub> is used as a raw material for the production of agricultural fertilizer,<sup>2</sup> which is a key to food security. Small fractions of NH<sub>3</sub> are used as a refrigerant, cleaning agent, and pharmaceutical feedstock.<sup>3</sup> In addition, NH<sub>3</sub> has recently re-emerged as an important and carbon-free energy carrier due to its advantages including high energy density (3 kW h kg<sup>−1</sup>), high hydrogen content (~18 wt%), facile storage and transportation, and the infrastructure maturity.<sup>4,5</sup>

Currently, the industrial production of NH<sub>3</sub> still relies on the Haber–Bosch process, a technology invented over a century ago.<sup>6</sup> Hydrocarbon steam reforming of high emissions is often utilized to provide H<sub>2</sub>, although renewable H<sub>2</sub> from water electrolysis is under heavy investigation as a new supply.<sup>1,7,8</sup> The Haber–Bosch process combines H<sub>2</sub> with

N<sub>2</sub> separated from air in a high-temperature (400–600 °C) and high-pressure (100–300 bar) reactor containing iron-based catalysts to produce NH<sub>3</sub>. The reaction is expressed as



Nevertheless, the conventional Haber–Bosch process suffers from various drawbacks. For example, it consumes 1–2% of the global energy supply, and it is responsible for approximately 1.2% of global CO<sub>2</sub> emissions.<sup>9,10</sup> Moreover, the energy cost of transporting fertilizer to remote areas needs to be considered because of the centralization of large-scale Haber–Bosch plants. To reach global carbon neutrality and align with the United Nations' Sustainable Development Goals,<sup>11</sup> there is an urgent need to develop novel, energy-efficient, and environmentally friendly technologies for distributed NH<sub>3</sub> synthesis.

Extensive studies have been conducted over the past decades with the goal of producing NH<sub>3</sub> under milder conditions. Compared with first-generation iron-based catalysts, ruthenium-based catalysts synthesize NH<sub>3</sub> at lower temperatures (300–450 °C) and pressures (40–150 bar), but their applications are constrained by the high cost and severe H<sub>2</sub> poisoning.<sup>5,12–14</sup> To improve the catalytic performance of transition metals, it is important to modify the nitrogen

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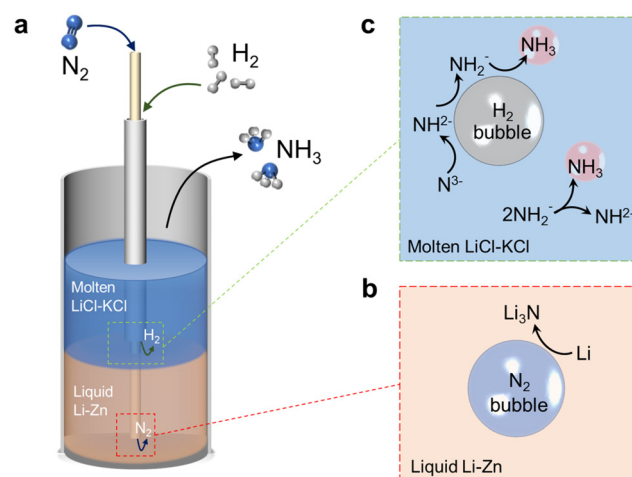
adsorption energy or even circumvent the scaling relation exerted on them.<sup>15–24</sup> For example, Wang *et al.* employed LiH as the second active center for hydrogenation to intervene in TM-mediated catalysis and achieved high catalytic activities at only 150–350 °C and 1 bar.<sup>16</sup> Peng *et al.* have reported a RuCo dual single-atom active catalyst to decouple the sites for N<sub>2</sub> activation from those for the desorption of N-containing intermediate species, and realized a high synthesis rate of NH<sub>3</sub> at merely 200 °C and 10 bar.<sup>19</sup> In addition, the ternary intermetallic electride LaCoSi and the related dual-site mechanism studies have been reported for effective ammonia synthesis with N<sub>2</sub> dissociation on the Co site and N hydrogenation on the La site, circumventing the scaling relation.<sup>25–28</sup>

The search for non-transition metal catalysts with high activity, stability, and cost effectiveness represents another important research direction. Kawamura and Taniguchi have reported a route using sodium (Na) melt as an effective catalyst for NH<sub>3</sub> synthesis from N<sub>2</sub> and H<sub>2</sub> at 500–590 °C under atmospheric pressure.<sup>29</sup> The function of the liquid Na was to ionize N<sub>2</sub> and dissolve nitrogen.<sup>30</sup> In addition, Li has been reported to spontaneously split the triple bond in N<sub>2</sub> to form Li<sub>3</sub>N under ambient conditions.<sup>31</sup> Li<sub>3</sub>N can be protonated by H<sub>2</sub>O to form NH<sub>3</sub> and LiOH; the LiOH product can be electrolyzed to regenerate Li, closing the loop.<sup>32</sup> However, the step of LiOH electrolysis remains a challenge mainly due to the consumption of Li by LiOH.<sup>33,34</sup> Solid Li-based alloys have also gained significant research attention for thermocatalytic NH<sub>3</sub> synthesis.<sup>35,36</sup> For example, Yamaguchi *et al.* reported NH<sub>3</sub> synthesis from N<sub>2</sub> and H<sub>2</sub> using a solid Li–Sn alloy below 400 °C under ambient pressure.<sup>37</sup> However, the solid Li–Sn pseudo-catalyst suffered from performance degradation after the first cycle. Meanwhile, the hydrogenation of the intermediates (*e.g.*, LiNH<sub>2</sub>, Li<sub>2</sub>NH) was limited by both thermodynamics and kinetics.<sup>38</sup> The reaction kinetics between Li<sub>2</sub>NH and H<sub>2</sub> for NH<sub>3</sub> synthesis might be improved by the addition of Li<sub>2</sub>O as a scaffold material.<sup>39</sup> Recently, Tang *et al.* reported an innovative Li-based loop for NH<sub>3</sub> synthesis in a liquid catalytic system comprising a Li–Sn alloy and LiCl–KCl salt.<sup>40</sup> The materials used were earth abundant and did not need complicated preparation. The liquid alloy and molten salt self-segregated by density into two distinct layers due to their immiscibility.<sup>33,41</sup> The Li-based loop consisted of three simultaneous steps: (i) N<sub>2</sub> fixation by Li in the liquid alloy to produce Li<sub>3</sub>N, (ii) the hydrogenation of Li<sub>3</sub>N dissolved in the molten salt to form NH<sub>3</sub> and LiH, and (iii) LiH decomposition to regenerate Li. The liquid catalytic system took advantage of the floating and dissolution of Li<sub>3</sub>N into the molten salt and circumvented the scaling relation exerted on Li, leading to effective NH<sub>3</sub> synthesis. In place of alkali metals, alkaline earth metals might also be candidate materials for fixing N<sub>2</sub> due to their high reactivity and low cost.<sup>32,40,42</sup> However, the mechanisms for nitrogen fixation and ammonia synthesis remained elusive, and the operation temperature of the liquid alloy–salt catalytic systems reported

was relatively high (*e.g.*, 510 °C for the system with a Li–Sn liquid alloy and LiCl–KCl molten salt). For a fixed pressure, lowering the operating temperature is critically important for increasing the equilibrium molar fraction of the NH<sub>3</sub> product based on the thermodynamic analysis.<sup>40,43</sup> The goal of unlocking the full potential of the liquid alloy–salt catalytic systems for NH<sub>3</sub> synthesis motivates the investigations, particularly unveiling the mechanisms of NH<sub>3</sub> formation and release.

It is necessary to identify the ideal materials for the liquid alloy–salt catalytic systems to achieve high performance. Based on the previous work,<sup>40</sup> several criteria are proposed here, including: (i) spontaneous reaction of N<sub>2</sub> with the reactive metal to form a metal nitride when the alloy (or metal) is exposed to N<sub>2</sub>, (ii) facile transport (*e.g.*, flotation, diffusion) of the metal nitride from the alloy (or metal) to the salt, (iii) easy dissolution and ionization of the metal nitride in the salt, (iv) spontaneous reaction between the nitride ion and the proton source to produce NH<sub>3</sub>; a metal hydride or salt is also formed depending on the type of proton source, (v) the easy release or collection of NH<sub>3</sub> from the catalytic system, and (vi) effective conversion of the metal hydride or salt formed in (iv) back to the reactive metal. According to the phase diagram,<sup>44</sup> the Li–Zn alloy of appropriate composition remains liquid at relatively low temperatures (Fig. S1a†). In addition, Li–Zn has a greater Li activity than Li–Sn for a fixed Li content,<sup>45</sup> indicating a greater capacity for N<sub>2</sub> fixation. Based on these properties, the liquid Li–Zn is investigated as a promising alloy candidate in the liquid catalytic system for operating at lower temperatures.

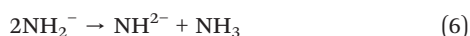
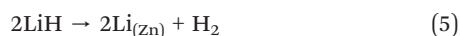
Fig. 1a shows a schematic of the liquid alloy–salt catalytic system. N<sub>2</sub> and H<sub>2</sub> are fed through the bubbling tubes to the



**Fig. 1** (a) Schematic illustration of the liquid catalytic system comprising the Li–Zn alloy and LiCl–KCl eutectic salt for NH<sub>3</sub> synthesis from N<sub>2</sub> and H<sub>2</sub> inside an alumina bubble-column reactor. (b) N<sub>2</sub> fixation by Li in the liquid Li–Zn alloy to form Li<sub>3</sub>N at the N<sub>2</sub> bubble/alloy interface. (c) The N<sub>3</sub><sup>-</sup> ions in the molten salt react with H<sub>2</sub> to produce NH<sub>2</sub><sup>-</sup>, NH<sub>2</sub><sup>-</sup> and NH<sub>3</sub>, successively, at the H<sub>2</sub> bubble/salt interface. Additionally, the bimolecular decomposition of NH<sub>2</sub><sup>-</sup> for producing NH<sub>3</sub> and NH<sub>2</sub><sup>-</sup> is also shown.



bottoms of the Li–Zn liquid alloy and LiCl–KCl molten salt, respectively.  $N_2$  is fixed by Li in the Li–Zn liquid alloy to form  $Li_3N$  (reaction (2) and Fig. 1b). Driven by the buoyancy and also the rising bubbles,  $Li_3N$  floats up to the molten salt and dissolves into  $Li^+$  and  $N^{3-}$  because the highly ionic molten salt has enough attraction to break down the  $Li_3N$  compound (reaction (3)).<sup>46,47</sup> The  $N^{3-}$  ions react with  $H_2$  to form  $NH_3$  and  $H^-$  (reaction (4) and Fig. 1c). Reaction (4) occurs stepwise *via* the formation of the imide ion ( $NH_2^-$ ) and the amide ion ( $NH_2^-$ ), as expressed by reactions (4a)–(4c).<sup>48</sup> In the presence of Zn, a small portion of the LiH product (a combination of  $Li^+$  and  $H^-$ ) might decompose to regenerate Li that alloys with Zn at the alloy/salt interface (reaction (5)).<sup>40,49,50</sup> In addition,  $NH_3$  can also be produced by bimolecular decomposition of  $NH_2^-$  (reaction (6) and Fig. 1c).<sup>48</sup> The standard molar Gibbs free energy change of the relevant reactions in the Li–N–H system are shown in Fig. S1b.† We systematically study the roles of multiple variables in nitrogen fixation and ammonia synthesis, and gain a deeper understanding of the process through combined experiments and modeling. The catalytic system integrates the functions of the Li–Zn liquid alloy and the LiCl–KCl molten salt, and is demonstrated for effective  $NH_3$  synthesis at relatively low temperatures and ambient pressure.



## 2. Experimental section

### 2.1. Pretreatment of the Li–Zn mixture and the LiCl–KCl eutectic salt

The volume expansion caused by the Li–Zn alloying is relatively small.<sup>51</sup> However, considering the large difference in the melting point and the density between Li and Zn, solid Li and Zn were pretreated to prepare the Li–Zn mixture, which would make the complete alloying easier in the subsequent high-temperature experiments. The desired amounts of Li strips and Zn granules were mixed in a small alumina crucible (OD: 14 mm, ID: 12 mm, height: 44 mm) that was placed in a bigger alumina container (OD: 80 mm, ID: 76 mm, height: 100 mm) with an alumina cover. The

system was heated at 260 °C for 2 hours on a hot plate (IKA, RCT Basic) inside an Ar-filled glovebox. The phases of the pre-treated Li–Zn mixture were determined to be LiZn and Li by X-ray diffraction (Fig. S3†).

To remove the moisture, the LiCl or KCl powders were transferred to an alumina crucible (OD: 80 mm, ID: 76 mm, height: 100 mm) in an electric furnace, and heated at 300 °C for 12 hours under vacuum, during which Ar was flowed into the system at 20 cm<sup>3</sup> min<sup>−1</sup>. The pre-treated salt was collected and stored in the Ar-filled glovebox. LiCl and KCl were weighted with a molar ratio of 0.592:0.408 and fully mixed using a vortex mixer (Vortex-Genie 2, Scientific Industries, Inc.) for 10 minutes to obtain the LiCl–KCl eutectic mixture. The actual amounts of metals and salts used for different experimental conditions are shown in Table S2.†

### 2.2. Bubble column reactors for $NH_3$ synthesis

Bubble column reactors were assembled using one-end-closed alumina tubes (OD: 18 mm; ID: 13 mm; height: 400 mm) and Swagelok Ultra-Torr™ vacuum fittings, as shown in Fig. S2.† The pretreated Li–Zn mixture and LiCl–KCl eutectic mixture were transferred sequentially to the alumina tube and sealed by vacuum fittings. The assembly was done in an Ar-filled glovebox with  $O_2$  and  $H_2O$  levels less than 10 ppm.

Inside a vertical electric furnace, the reactor fed with Ar (20 cm<sup>3</sup> min<sup>−1</sup>) was first maintained at 300 °C for 12 hours under vacuum to remove the moisture. Afterwards, the temperature was ramped to 500 °C, maintained for 30 min, and then cooled to 400 °C under Ar atmosphere to obtain the Li–Zn liquid alloy (height: ~5 cm) and LiCl–KCl eutectic molten salt (height: ~5 cm). The density of the 70 mol% Li–Zn alloy (2.33 g cm<sup>−3</sup>) is greater than that of LiCl–KCl eutectic salt (1.69 g cm<sup>−3</sup>) at 400 °C,<sup>52–54</sup> so the alloy was below the salt. Then, pure  $N_2$  (5 cm<sup>3</sup> min<sup>−1</sup>) and  $H_2$  (15 cm<sup>3</sup> min<sup>−1</sup>) were introduced into the reactor in either a separate or mixed way. The flow rates of  $N_2$  and  $H_2$  were regulated using mass flow controllers (MFC; Omega Engineering Inc., Part No.: FMA5508A).

The  $NH_3$  product in the effluent gas was absorbed by 0.05 mol L<sup>−1</sup>  $H_2SO_4$  aqueous solution (10 mL) which was replaced every two hours to determine the  $NH_3$  synthesis rate. A colorimetric indicator method was coupled with ultraviolet visible spectroscopy (UV-Vis) for quantification of  $NH_3$  trapped in the  $H_2SO_4$  aqueous solution based on the  $NH_3$  standard calibration curve.<sup>40</sup> An Agilent Cary 5000 UV-Vis–NIR spectrophotometer was used to obtain the relationship between the absorbance and the wavelength.

### 2.3. Material characterization

X-ray diffraction (XRD) with Cu K $\alpha$  radiation and a  $\theta$ – $2\theta$  diffraction geometry was used to determine the crystalline phases of the solidified alloy and salt (Bruker, D8 Advance, USA). The crystalline phases were identified with the International Centre for Diffraction Data (ICDD) powder diffraction file (PDF)-4+ database. The sample preparation



was carried out in an Ar-filled glovebox, and polyimide tape was used to seal the sample to prevent oxidation during the sample transfer and XRD measurement. Chemical elements on the alloy or salt surface were analyzed using X-ray photoelectron spectroscopy (XPS) with an ESCALAB™ 250Xi XPS microprobe (Thermo Fisher Scientific, USA). The NIST XPS database was used for chemical state identification.<sup>55</sup> All binding energies were calibrated by referencing to the C 1s peak of adventitious carbon at 284.8 eV, similar to previous reports.<sup>56,57</sup> The sample was prepared in an Ar-filled glovebox and then transferred to the XPS sample chamber using a portable sample transfer-bin.

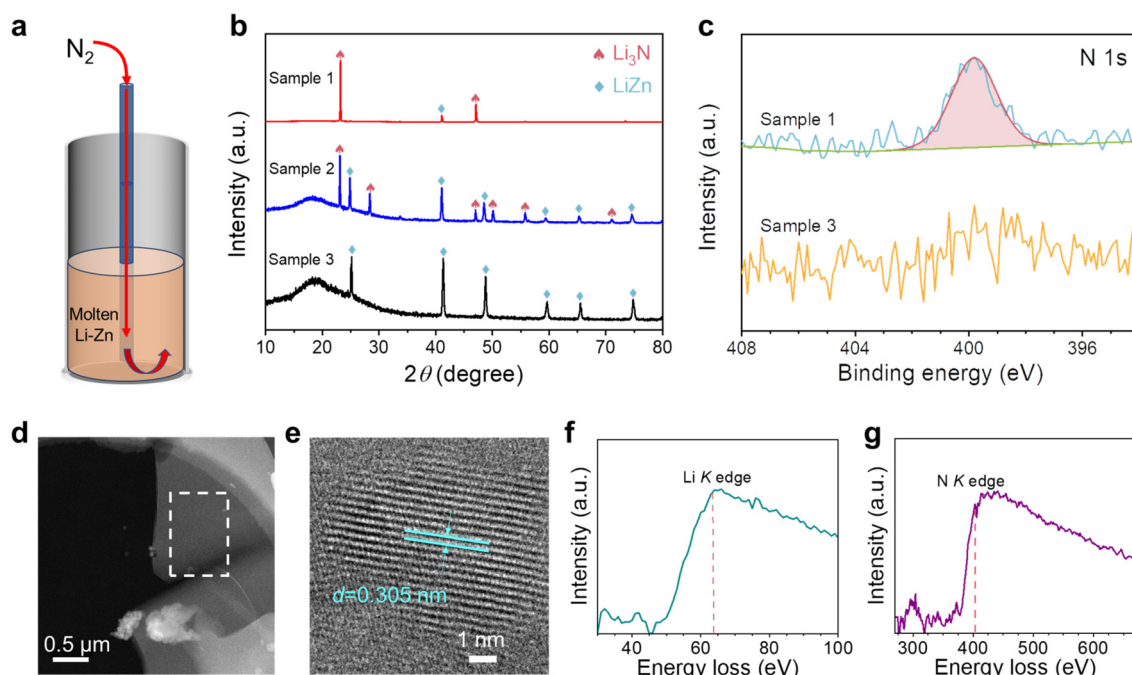
Raman spectroscopy was used to characterize the samples with an Andor Shamrock SR-500i Raman system ( $\times 50$  objective lens,  $\times 600$  grating, and 532 nm laser excitation). The surface morphologies and elemental distribution were studied using scanning electron microscopy (SEM) and energy dispersive X-ray spectroscopy (EDS) on a JEOL SEM IT-500HR/LA. The transmission electron microscopy (TEM) images, scanning transmission electron microscopy (STEM) images and electron energy-loss spectroscopy (EELS) results were collected using a 300 kV aberration-corrected transmission electron microscope (JEOL GrandARM-300F) equipped with a double spherical corrector with a Gatan Oneview IS camera. The sample was first ground with a pestle and mortar into powders, some of which were transferred to a copper grid and then fixed in a holder for

characterization. The sample preparation was all done inside an Ar-filled glovebox.

### 3. Results and discussion

#### 3.1. N<sub>2</sub> fixation by the Li–Zn liquid alloy

**3.1.1. Characterization of the solidified alloy column after the reaction.** N<sub>2</sub> activation and fixation is the critical step for NH<sub>3</sub> synthesis in the liquid alloy–salt catalytic system, considering the strong N≡N triple bond.<sup>40</sup> We first investigated the reaction of N<sub>2</sub> with the Li–Zn alloy inside a bubble column reactor as illustrated in Fig. 2a. The pre-treated 70 mol% Li–Zn mixture inside the reactor was heated to 500 °C and then cooled to operation temperature (400 °C) under Ar atmosphere to obtain the liquid alloy with a height of approximately 5 cm. Then, N<sub>2</sub> was introduced through an alumina bubbling tube to the liquid alloy bottom at 5 cm<sup>3</sup> min<sup>−1</sup> for 1 hour. The reactor was subsequently cooled to room temperature and dismantled for material characterization. Fig. 2b shows the XRD patterns of the samples collected from the solidified alloy column. Sample 1 was collected from the wall close to the top of the alloy column, sample 2 was collected right from the top, whereas sample 3 was collected from the lower portion. Based on the XRD pattern for sample 1, the dominant phase was identified to be high-crystallinity  $\alpha$ -Li<sub>3</sub>N with preferred orientations and a small amount of the LiZn intermetallic phase.<sup>58</sup> No



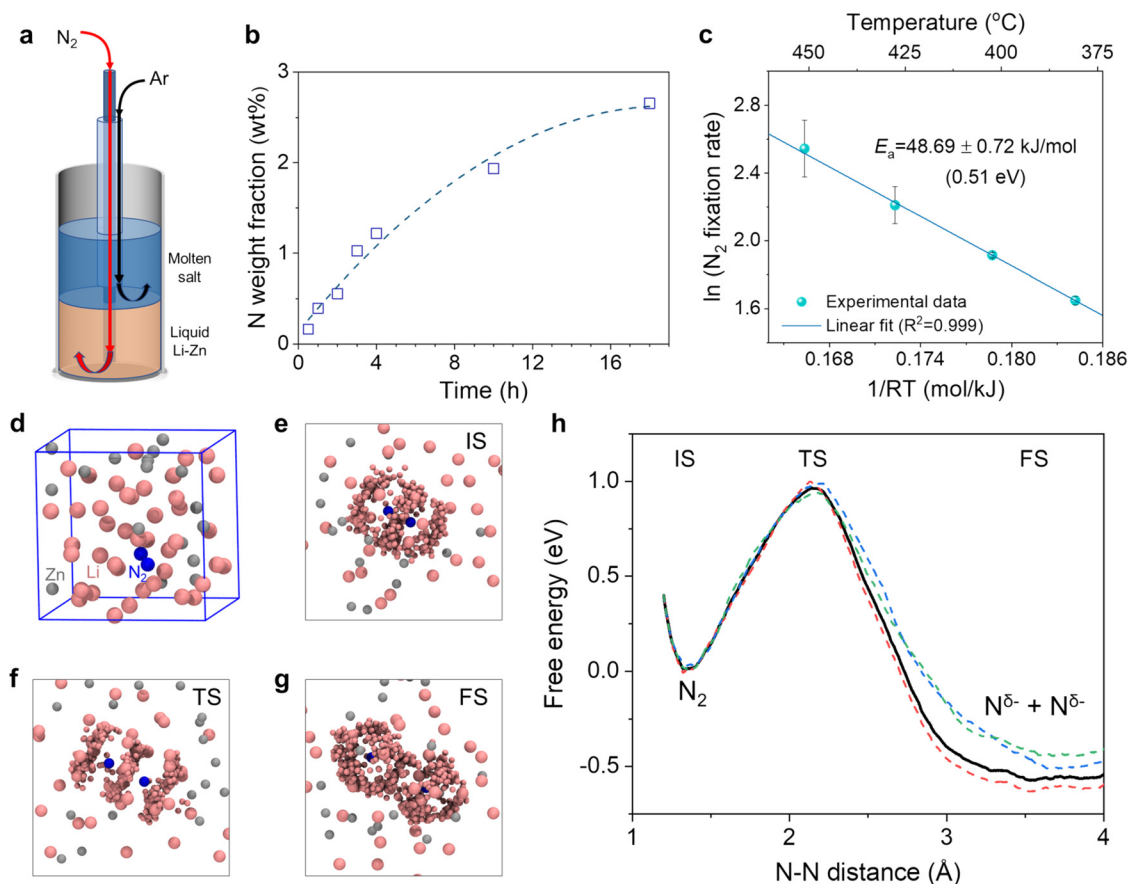
**Fig. 2** (a) Schematic of the reactor containing the liquid Li–Zn alloy (height: ~5 cm) bubbled with N<sub>2</sub> (5 cm<sup>3</sup> min<sup>−1</sup>). (b) XRD patterns of the three samples collected from different places of the solidified alloy column. The characteristic peaks for  $\alpha$ -Li<sub>3</sub>N (PDF No. 76-0820) and LiZn intermetallic (PDF No. 03-0954) were indexed. The wide signal at ~18° observed for samples 2 and 3 was attributed to the polyimide film cover. (c) High-resolution N 1s XPS results of sample 1 and sample 3. (d) STEM image for sample 1 on a copper grid. (e) High-resolution TEM image of sample 1 with the (100) interplanar spacing identified to be 0.305 nm. (f) The Li K-edge and (g) the N K-edge EELS spectra for the area enclosed by the white dashed rectangle in (d).



characteristic peaks for  $\text{LiZnN}$  or  $\text{Zn}_3\text{N}_2$  were observed.<sup>47,59</sup> The presence of  $\alpha\text{-Li}_3\text{N}$  was consistent with the XPS characterization results (Fig. 2c),<sup>60,61</sup> validating reaction (2). The XRD pattern for sample 2 shows the diffraction peaks for both  $\alpha\text{-Li}_3\text{N}$  and  $\text{LiZn}$ ,<sup>58</sup> yet, it's noted that the crystallinity seemed not as high as that in sample 1. The XRD results for sample 2 were consistent with the SEM and EDS results (Fig. S4†). The XRD pattern for sample 3 didn't show the diffraction peaks for  $\text{Li}_3\text{N}$ ; the N 1s XPS result in Fig. 2c didn't exhibit the characteristic peaks either, indicating the absence of  $\text{Li}_3\text{N}$ .  $\text{Li}_3\text{N}$  (density:  $1.296 \text{ g cm}^{-3}$  at room temperature)<sup>62</sup> floated to the top of the liquid alloy due to the density difference and phase immiscibility.<sup>40</sup> It's noted that, based on the Li-Zn phase diagram<sup>44</sup> (Fig. S1a†), the cooling of the 70 mol% Li-Zn alloy would transform to a mixture of  $\text{LiZn}$  and  $\text{Li}$ . However, the amount of  $\text{Li}$  was below the XRD detection limit because  $\text{N}_2$  fixation consumed some  $\text{Li}$  in the Li-Zn alloy.

To analyze the local structure and elemental composition, scanning transmission electron microscopy (STEM) and electron energy-loss spectroscopy (EELS) were used to characterize sample 1. The STEM images show the particles of sample 1 after grinding (Fig. 2d and S5†). Fig. 2e shows the lattice-resolved TEM image of sample 1, and the interplanar spacing ( $d$ ) of 0.305 nm was indexed to the (100) planes of  $\alpha\text{-Li}_3\text{N}$ . Fig. 2f and g confirmed the presence of  $\text{Li}$  and  $\text{N}$  based on their K-shell ionization edges at 63 eV and 403 eV, respectively.<sup>63,64</sup>

**3.1.2. Kinetic parameter measurements for  $\text{N}_2$  fixation by the Li-Zn liquid alloy.** To evaluate the time dependence of  $\text{N}_2$  fixation, we performed a series of experiments for different durations at 400 °C. To quantify the amount of nitrogen fixed by the liquid Li-Zn alloy, we proposed the use of a layer of molten  $\text{LiCl-KCl}$  (height: 5 cm; mass: 8.8 g) on top of the liquid Li-Zn to collect  $\text{Li}_3\text{N}$  through dissolution (Fig. 3a).  $\text{N}_2$  ( $5 \text{ cm}^3 \text{ min}^{-1}$ ) and  $\text{Ar}$  ( $15 \text{ cm}^3 \text{ min}^{-1}$ ) were flowed into the



**Fig. 3** Kinetics and AIMD simulation for  $\text{N}_2$  fixation by the Li-Zn liquid alloy. (a) Schematic diagram of the reactor containing 70 mol% Li-Zn alloy (height: 5 cm) in the bottom bubbled with  $\text{N}_2$  ( $5 \text{ cm}^3 \text{ min}^{-1}$ ) and the eutectic  $\text{LiCl-KCl}$  molten salt (height: 5 cm) on the top bubbled with  $\text{Ar}$  ( $15 \text{ cm}^3 \text{ min}^{-1}$ ). (b) Dependence of the nitrogen weight fraction in the salt on the reaction time. The dashed line served as a guide for the eyes. (c) Arrhenius plot for the  $\text{N}_2$  fixation rate in the temperature range of 380–450 °C. (d) Snapshot of the model system after a pre-relax simulation for 5 ps. The cubic box represents the periodic unit cell of the infinite bulk system. Color code: pink, Li; gray, Zn; blue, N. (e–g) The local environments of the initial, transition, and final states (IS, TS, and FS) during  $\text{N}_2$  cleavage, respectively. The statistical distribution of Li and Zn near the two nitrogen atoms was illustrated by superimposed snapshots with smaller spheres. (h) The free energy profile as a function of the N-N distance during the  $\text{N}_2$  cleavage process, as denoted by the black bold line. The work profiles of 3 slow-growth trajectories are also shown, as denoted by the dashed lines in color.





liquid alloy and the molten salt, respectively. At the end of each experiment, Ar bubbling was kept for an extra period (30 min) to homogenize the distribution of  $\text{Li}_3\text{N}$ . Afterwards, the reactor was cooled and disassembled for material characterization. After the experiments with  $\text{N}_2$  flow for 2 hours and 18 hours, alloy samples were collected from the upper, middle and lower parts along the solidified Li–Zn columns, and characterized by XPS. Their N 1s XPS signals were all below the detection limit (Fig. S6†), confirming that  $\text{Li}_3\text{N}$  floated up and dissolved into the molten salt on top of the alloy. To further quantify the nitrogen content, three pieces of solidified alloy (0.1165 g, 0.0998 g, and 0.1522 g) were sampled, and each was placed into 10 mL of 0.05 mol  $\text{L}^{-1}$   $\text{H}_2\text{SO}_4$  aqueous solution to dissolve any  $\text{Li}_3\text{N}$ . Based on the UV-Vis measurements of the solutions, the nitrogen weight fractions in those three solidified alloy samples were determined to be 0.039%, 0.051%, and 0.011%, respectively; these small values were consistent with the absence of N 1s XPS peaks in Fig. S6†. The result demonstrated the validity of using the molten LiCl–KCl to collect and quantify the amount of  $\text{Li}_3\text{N}$  fixed.

Fig. 3b shows the nitrogen weight fraction in the salt by analyzing five solidified samples from each experiment and then taking the mass-weighted average (see detailed data in Table S3†). In each experiment, the relatively homogeneous distribution of nitrogen in the solidified salt was consistent with the dissolution and ionization of  $\text{Li}_3\text{N}$  in the molten salt (reaction (3)). The tiny amount of nitrogen in the impurities of the commercial LiCl and KCl was subtracted (Fig. S7†).<sup>65</sup> The N weight fraction in the salt increased with the reaction time (Fig. 3b). Yet, the increase gradually slowed down, likely because the reaction of Li with  $\text{N}_2$  reduced the Li molar content in the alloy (e.g., from 70% to 66.81% for the 18-hour experiment). The Li activity declined exponentially with the decreasing content,<sup>45</sup> and accordingly the ability of the liquid alloy for  $\text{N}_2$  fixation gradually became weaker. In addition, the Li content decrease shifts the alloy composition towards the liquidus line based on the Li–Zn phase diagram (Fig. S1a†) and might change the physicochemical properties (e.g., viscosity and surface tension) of the liquid alloy. Besides, foaming and channel formation in the liquid alloy could also be possible. Their effects on  $\text{N}_2$  fixation will be investigated in future studies.

To evaluate the temperature dependence of the  $\text{N}_2$  fixation rate, we performed a series of experiments at different temperatures from 380 to 450 °C using the setup shown in Fig. 3a. Each experiment last for 2 hours. The nitrogen fixation rate by the 70 mol% Li–Zn liquid alloy was calculated by dividing the total amount of  $\text{Li}_3\text{N}$  in the salt above by the reaction time (Note S1 and Fig. S8†). Fig. 3c shows the Arrhenius plot of the natural logarithm of the  $\text{N}_2$  fixation rate vs.  $1/RT$ , where  $R$  is the gas constant ( $8.31 \text{ J K}^{-1} \text{ mol}^{-1}$ ). The value of the apparent activation energy ( $E_a$ ) for  $\text{N}_2$  fixation by the liquid Li–Zn was determined to be approximately  $48.69 \text{ kJ mol}^{-1}$ , equivalent to 0.51 eV. Such a low energy barrier was mainly attributed to the high activity of Li for  $\text{N}_2$  activation

and fixation,<sup>35</sup> as the overlap between the Zn d orbital and the Li s orbital was relatively small,<sup>66</sup> as will be discussed in section 3.1.3. To investigate the contribution of Zn to  $\text{N}_2$  fixation and the subsequent  $\text{NH}_3$  synthesis, we performed another experiment with a liquid catalytic system comprising pure Zn (height: 5 cm) and eutectic molten LiCl–KCl on the top (height: 5 cm) at 500 °C (Fig. S9†). The results suggest that the liquid Zn itself has negligible activity for  $\text{N}_2$  fixation.

**3.1.3. Understanding  $\text{N}_2$  fixation by Li–Zn from AIMD simulations.** To gain insights into  $\text{N}_2$  fixation by the Li–Zn liquid alloy, we investigated this reaction by combining density functional theory (DFT)-based *ab initio* molecular dynamics (AIMD) simulations with the free energy sampling method. From the aspect of theoretical modeling, one important obstacle in understanding the reaction mechanism is the accurate description of the random nature and the complex and dynamic structure of the melts. The AIMD simulation is a straightforward method to treat the movement of liquid species at finite temperatures.<sup>56,67</sup> The computational details are provided in Note S7†.

To examine the reactivity of  $\text{N}_2$  cleavage in liquid 70 mol% Li–Zn alloy, a randomly mixed  $\text{Li}_{42}\text{Zn}_{18}$  model was built in a periodic cubic cell with the edge length = 11.2 Å. Then, a  $\text{N}_2$  molecule was placed in the cell, and a pre-relax simulation was performed for 5 ps, allowing the liquid catalyst around the  $\text{N}_2$  molecule to find the favorable arrangements. Fig. 3d shows the structure obtained, suggesting that the distributions of the Li and Zn atoms were not entirely random after the pre-relax simulation (Fig. S10a and b†). Further AIMD simulations were performed under the experimental condition of 400 °C. Fig. 3h shows the free energy change for  $\text{N}_2$  cleavage to two  $\text{N}^{\delta-}$  as a function of the N–N distance. The value of  $\delta$  was calculated to be  $3.3 \pm 0.2$  based on the Bader charge analysis.<sup>68</sup> The activation free energy ( $G_a$ ) is defined as the energy difference between the transition state (TS) and the initial state (IS), and the reaction free energy ( $\Delta G$ ) is defined as the energy difference between the final state (FS) and the IS. The  $\text{N}_2$  cleavage in the liquid Li–Zn alloy showed a  $G_a$  of 1.0 eV and  $\Delta G$  of  $-0.5 \text{ eV}$ , which are much lower than the cleavage in the gas phase.<sup>69</sup> The results suggest that the  $\text{N}_2$  cleavage would be favorable in the liquid Li–Zn than in the gas. There is a discrepancy between the theoretical  $G_a$  and experimental values from the Arrhenius plot. The discrepancy may have two main origins: firstly, the slope of the Arrhenius plot is the activation enthalpy actually, rather than the activation free energy; and secondly, the dissolution of  $\text{N}_2$  in the liquid alloy was not taken into account in the theoretical model. In spite of this, the main catalytic mechanisms can still be captured through theoretical simulations. The local environment near  $\text{N}_2$  in the IS, TS and FS is illustrated in Fig. 3e–g, respectively. From these figures we found that, in each stage of  $\text{N}_2$  cleavage, only Li atoms in the Li–Zn alloy interact with the N atoms, while Zn atoms rarely appear in the vicinity of the N atom. The results indicate that the  $\text{N}_2$  cleavage in the Li–Zn liquid alloy is assisted by the Li atoms, and the Zn atoms mainly act as a

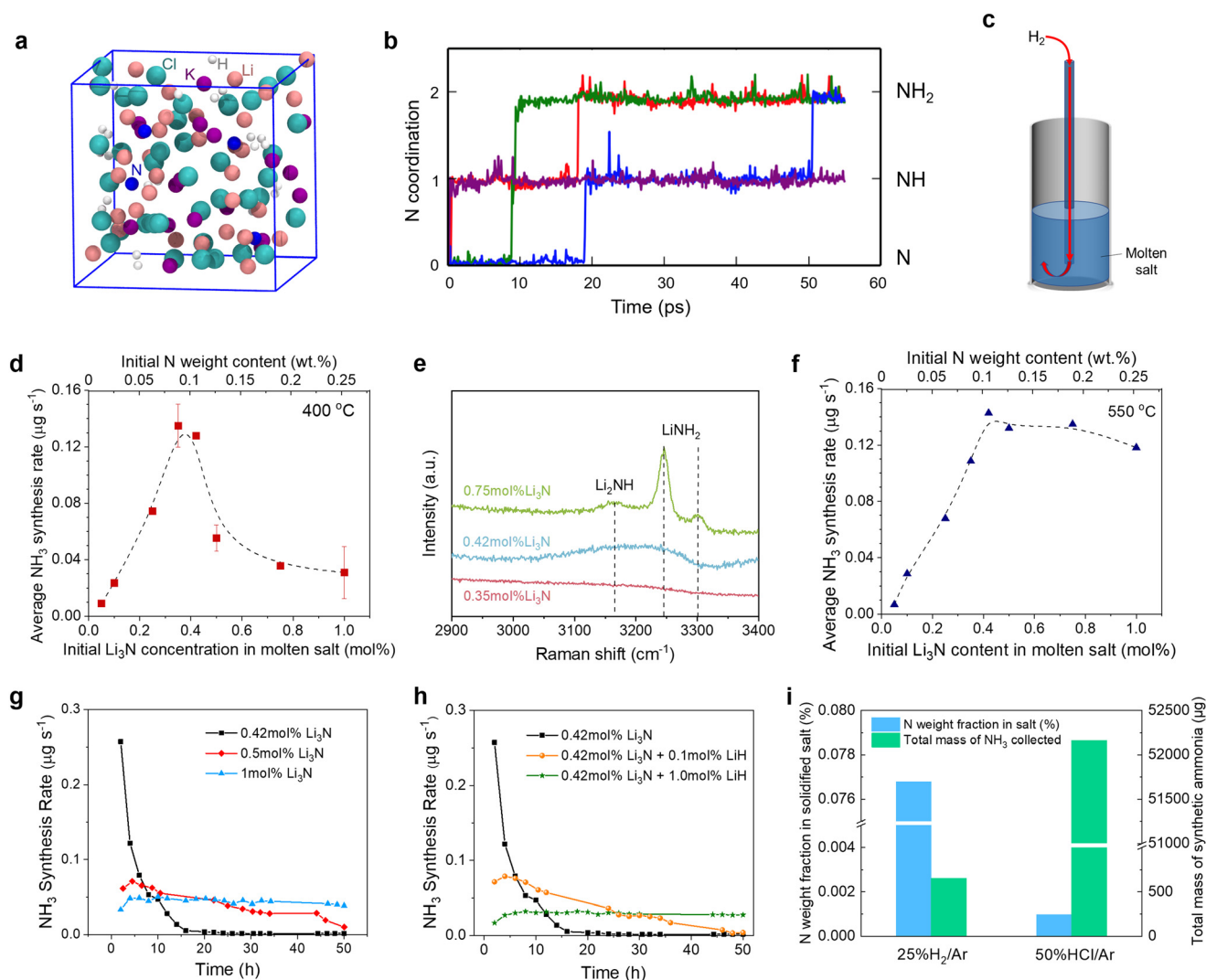


solvent for Li. Notably, Zn was added to the experimental system mainly based on the following considerations. First, the addition of Zn increases the density of the liquid alloy. Thus, the liquid alloy could stay beneath the molten salt (Fig. 1a and 3a), and the  $\text{Li}_3\text{N}$  formed by the reaction between  $\text{N}_2$  and Li in the alloy could float to the molten salt. Second, adding Zn can reduce the activity of Li, decreasing the corrosiveness to the alumina reactor.

To investigate the impact of Zn addition on the Li activity in the 70 mol% Li–Zn liquid alloy, we conducted an analysis of the projected density of states (PDOS) in the  $\text{Li}_{42}\text{Zn}_{18}$

model. The PDOS analysis (Fig. S10c and d†) indicates a reduced overlap of the d orbital of Zn and the s orbital of Li, consistent with previous findings,<sup>66</sup> despite the differences in their modeling system where the Li atoms were aggregated. These computational results demonstrate that in the Li–Zn system, the charge transfer from Zn to Li atoms is minimal, and as a result, the activity of Li is not significantly hindered for  $\text{N}_2$  fixation.

We also examined the reactivity of  $\text{N}_2$  cleavage in liquid pure Zn (Note S2, Fig. S10e and f†), liquid pure Li (Note S2, Fig. S10g and h†), and liquid 60 mol% Li–Sn alloy (Note S3



**Fig. 4** The reaction of  $\text{Li}_3\text{N}$  with  $\text{H}_2$  for  $\text{NH}_3$  synthesis in the molten salt. (a) A snapshot of the model system of  $\text{Li}_3\text{N}$  hydrogenation reaction in the molten  $\text{LiCl-KCl}$  at the end of the AIMD simulation. Color code: pink, Li; purple, K; cyan, Cl; blue, N; white, H. (b) The trajectory of the coordination number of each of the four N atoms in the AIMD simulation. Different colors were used for these trajectories. (c) Schematic diagram of the experimental setup for bubbling  $\text{H}_2$  ( $15 \text{ cm}^3 \text{ min}^{-1}$ ) to the eutectic  $\text{LiCl-KCl}$  molten salt (height: 5 cm) containing  $\text{Li}_3\text{N}$ . The dependence of the average  $\text{NH}_3$  synthesis rate on the  $\text{Li}_3\text{N}$  content in the molten salt in the experiments carried out for 8 hours at (d) 400 °C and (f) 550 °C. The dashed lines in (d) and (f) serve as guides for the eyes. (e) Raman spectra in the N–H stretching region of the solidified salt samples collected from the experiments that originally contained 0.35, 0.42, and 0.75 mol%  $\text{Li}_3\text{N}$  at 400 °C. The time dependence of the  $\text{NH}_3$  synthesis rate measured for the 50-hour experiments of bubbling  $\text{H}_2$  into (g) the eutectic  $\text{LiCl-KCl}$  molten salts containing varying amounts of  $\text{Li}_3\text{N}$  (0.42 mol%, 0.5 mol%, and 1.0 mol%) and (h) the eutectic  $\text{LiCl-KCl}$  molten salts containing 0.42 mol%  $\text{Li}_3\text{N}$  and varying amounts of  $\text{LiH}$  (0%, 0.1 mol%, and 1.0 mol%). (i) The N weight fractions in the solidified salts and the total amounts of  $\text{NH}_3$  collected in the experiments of feeding 25%  $\text{H}_2/\text{Ar}$  or 50%  $\text{HCl}/\text{Ar}$  to the melts containing 0.5 mol%  $\text{Li}_3\text{N}$  at 500 °C.





and Fig. S11†) with AIMD simulations. A comparison of the results suggests that the  $N_2$  cleavage in the pure liquid Zn is difficult due to the high activation energy of almost 4 eV. The liquid 70 mol% Li–Zn with a  $G_a$  of 1 eV is more reactive than the liquid 60 mol% Li–Sn with a  $G_a$  of 1.7 eV for  $N_2$  activation. The  $N_2$  cleavage in pure liquid Li shows a  $G_a$  of 0.8 eV, which is only 0.2 eV smaller than that in the 70 mol% Li–Zn. It can be inferred that the activity mainly originates from the high electron-donating ability of Li, and the 70 mol% Li–Zn liquid alloy still maintains the high activity of Li for  $N_2$  fixation.

### 3.2. The reaction of $Li_3N$ with $H_2$ for $NH_3$ synthesis in the molten salt

**3.2.1. Reactivity of  $Li_3N$  with  $H_2$  in the molten salt.** In the liquid alloy–salt catalytic system, the  $Li_3N$  formed in the Li–Zn liquid alloy would float up and dissolve into  $Li^+$  and  $N^{3-}$  in the molten salt. Previous studies have reported that the  $Li_3N$  dissolved in the molten salt promotes the spontaneous disproportionation of  $H_2$ , resulting in  $NH_3$  synthesis in the experiments at 450 and 510 °C.<sup>40,48</sup> The hydrogenation of  $N^{3-}$  dissolved in the molten salt has faster kinetics and a smaller activation barrier compared with that in the solid-state  $Li_3N$ .<sup>40,70,71</sup> To investigate the reactivity of  $Li_3N$  in the molten LiCl–KCl with  $H_2$ , we carried out AIMD simulation by building a randomly mixed system composed of 4  $Li_3N$  + 20 LiCl + 20 KCl in a periodic cubic cell of the edge length = 14.0 Å. Subsequently, 12  $H_2$  molecules were placed in the cell. Fig. 4a illustrates a snapshot of the system. To accelerate the reaction, the AIMD simulation was performed at an elevated temperature of 927 °C (1200 K). The basis for the use of elevated temperature is the assumption that the process shows Arrhenius behavior; *i.e.*, increasing temperature does not alter the pathway but only the rate.

From Fig. 4b, it is obvious that the two  $N^{3-}$  ions were readily hydrogenated to  $NH_2^-$  within only hundreds of femtoseconds (fs), suggesting that the models composed of  $N^{3-}$  and  $H_2$  were quite reactive. After 55 picoseconds (ps) of the AIMD simulation, the formation of three  $NH_2^-$  and one  $NH_2^-$  was observed, as shown in Fig. 4a and b. No  $NH_3$  was observed in the simulation, probably because the simulation time was still shorter than the experimental condition. The hydrogenation to  $NH_3$  may need a much longer time which is beyond the computational capacities. Moreover, no dehydrogenation processes occurred in the simulations. Therefore, we predict that the hydrogenation of  $Li_3N$  in the molten salt is reactive under the experimental conditions, and occurs stepwise as described by reactions (4a)–(4c). It's also noted that, through simulation, we found that some nitrogen ions are bound to the surrounding Li ions, making it difficult for these nitrogen ions to come into contact and react with  $H_2$ . The results suggest that it's vital to separate the  $Li^+$  and the nitrogen ions dissolved in the molten salt to accelerate  $NH_3$  synthesis.

**3.2.2. The effect of  $Li_3N$  concentration in the molten salt on  $NH_3$  synthesis.** Since the Li–Zn alloy has a strong ability for  $N_2$  activation and fixation, a significant amount of  $Li_3N$  could be formed in the alloy and then dissolved as  $Li^+$  and  $N^{3-}$  in the molten LiCl–KCl (Fig. 3b). The  $N^{3-}$  ions can react with the proton source such as  $H_2$  to produce  $NH_3$  (reaction (4)). Nevertheless, the excessive  $N^{3-}$  along with the chloride salts<sup>72</sup> might absorb the  $NH_3$  produced and suppress its release (reaction (7)).<sup>48,73</sup> As inspired by Sabatier's principle in catalysis,<sup>74</sup> we hypothesize that the  $N^{3-}$  concentration in the molten salt should be neither too high nor too small, and the optimal  $NH_3$  yield occurs when the  $N^{3-}$  concentration in the molten salt is of an intermediate value.



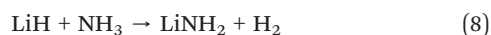
To test the hypothesis, we performed experiments to investigate the effect of the  $Li_3N$  concentration on  $NH_3$  synthesis at 400 °C in bubble column reactors (Fig. 4c). In each experiment, the LiCl–KCl eutectic molten salt (height: 5 cm) initially contained the desired amount of  $Li_3N$  varying from 0 to 1 mol%, and  $H_2$  was bubbled into the molten salt at 15 cm<sup>3</sup> min<sup>−1</sup> for 8 hours. The  $Li_3N$  used was synthesized by reacting pure Li tablets with  $N_2$  at 100 °C (Note S4 and Fig. S12†). The  $Li_3N$  concentrations in the all the experiments were below its solubility limit in the salt.<sup>46,47</sup> Fig. 4d shows a typical volcano plot. The average  $NH_3$  synthesis rate first increased and then decreased with the  $Li_3N$  concentration; *i.e.*, there was an optimum value for the  $Li_3N$  concentration in the molten salt for  $NH_3$  synthesis. Such behavior was attributed to the aforementioned competing factors including  $NH_3$  formation and absorption (reactions (4) and (7)). For a very low  $Li_3N$  concentration, the average  $NH_3$  synthesis rate was limited by the availability of  $N^{3-}$  at the salt/bubble interface to react with  $H_2$ . As the  $Li_3N$  concentration increased, reaction (4) shifted to the right and favored  $NH_3$  formation. Nevertheless, as the  $Li_3N$  concentration increased beyond approximately 0.4 mol%, the enhancement of  $NH_3$  absorption (reaction (7)) exceeded the formation of  $NH_3$  (reaction (4)). The  $NH_3$  desorption started to become limiting, and therefore the average  $NH_3$  synthesis rate gradually decreased at 400 °C. It's noted that the composition of 0.4 mol%  $Li_3N$  in the eutectic LiCl–KCl corresponded to 0.1 wt% nitrogen in the salt. The results suggested that the  $N^{3-}$  concentration can modulate the absorption strength of the molten salt, a key factor for  $NH_3$  release. After cooling, the solidified salt samples were collected from the experiments originally with 0.35, 0.42, and 0.75 mol%  $Li_3N$ . To avoid the moisture, they were enclosed in sealed quartz cuvettes inside an Ar-filled glovebox before Raman spectroscopy analysis (Fig. 4e). Only the salt with the relatively high original amount of  $Li_3N$  (0.75 mol%) exhibited the characteristic peaks at 3245 and 3300 cm<sup>−1</sup> for  $LiNH_2$  and at 3165 cm<sup>−1</sup> for  $\beta$ - $Li_2NH$ ,<sup>75–78</sup> consistent with the strong  $NH_3$  absorption by the excessive  $Li_3N$  (reaction (7)).



Using the setup shown in Fig. 4c, we also investigated the relation between the average  $\text{NH}_3$  synthesis rate and the  $\text{Li}_3\text{N}$  concentration at 550 °C, a much higher temperature than the target operation temperature in this work (400 °C). The average  $\text{NH}_3$  synthesis rate vs. the  $\text{Li}_3\text{N}$  concentration curve obtained at 550 °C also exhibited a volcano-like plot (Fig. 4f); yet, the changing trend was not exactly the same as that at 400 °C. In the low  $\text{Li}_3\text{N}$  concentration regime (0–0.4 mol%), the  $\text{NH}_3$  synthesis rate for both temperatures was close. However, in the high  $\text{Li}_3\text{N}$  concentration regime (0.4–1 mol%), the average  $\text{NH}_3$  synthesis rate at 550 °C decreased only slightly due to the following reasons. Reaction (7) became less thermodynamically favorable as the temperature increased from 400 °C to 550 °C (Fig. S1b†). In addition, the increased temperature would increase the kinetics of reaction (4), although reaction (4) became slightly unfavorable. Overall, the temperature increase exhibited a promoting effect for  $\text{NH}_3$  synthesis in the high  $\text{Li}_3\text{N}$  concentration regime.

The time dependence of the  $\text{NH}_3$  synthesis rate was also investigated in 50-hour experiments with  $\text{H}_2$  (15  $\text{cm}^3 \text{min}^{-1}$ ) bubbled into the eutectic LiCl–KCl of different  $\text{Li}_3\text{N}$  concentrations (0.42, 0.5, and 1.0 mol%) at 400 °C and ambient pressure (Fig. 4g). In the experiment with 0.42 mol%  $\text{Li}_3\text{N}$ , the  $\text{NH}_3$  synthesis rate decreased rapidly over the first 10 hours but the decreasing trend gradually slowed down, because the consumption of  $\text{Li}_3\text{N}$  and the accumulation of LiH made reaction (4) more difficult to proceed.<sup>40</sup> In the experiment with the  $\text{Li}_3\text{N}$  concentration raised to 0.5 mol%, the  $\text{NH}_3$  synthesis rate increased in the beginning, reached a peak value (0.071  $\mu\text{g s}^{-1}$ ) at approximately 4.5 hours, and then gradually decreased. A delayed emergence of the peak for the  $\text{NH}_3$  synthesis rate was due to the absorption of  $\text{NH}_3$  (reaction (7)), consistent with previous studies.<sup>48,79</sup>

When the  $\text{Li}_3\text{N}$  concentration in the molten salt was further increased to 1.0 mol%, the  $\text{NH}_3$  synthesis rate initially increased during the first 4 hours and then remained at a relatively stable value (Fig. 4g), indicating a subtle balance between various reactions. First, the high  $\text{Li}_3\text{N}$  concentration not only favored  $\text{NH}_3$  production (reaction (4)) but also led to a strong absorption of  $\text{NH}_3$  (reaction (7)). In addition, based on reactions (4c) and (6), the accumulation of  $\text{LiNH}_2$  would increase the thermodynamic driving force to promote  $\text{NH}_3$  production (reaction (6) was validated as shown in Fig. S13†). Eventually, the reaction products in the molten salt might be transformed into  $\text{Li}_2\text{NH}$ ,  $\text{LiNH}_2$ , and LiH. Similar to  $\text{Li}_3\text{N}$ , LiH can absorb  $\text{NH}_3$  as expressed by reaction (8), which is invertible to reaction (4c),<sup>80</sup> although the thermodynamic driving force for  $\text{NH}_3$  absorption by LiH was smaller than that by  $\text{Li}_3\text{N}$  (Fig. S1b†).



To reveal the role of LiH in the  $\text{NH}_3$  synthesis, we performed another two experiments of bubbling  $\text{H}_2$  (15  $\text{cm}^3 \text{min}^{-1}$ ) into

LiCl–KCl melts containing 0.42 mol%  $\text{Li}_3\text{N}$ –0.1 mol% LiH and 0.42 mol%  $\text{Li}_3\text{N}$ –1 mol% LiH, respectively, at 400 °C. As shown in Fig. 4h, with the addition of more LiH, the changing trend of the curve was similar to that of increasing  $\text{Li}_3\text{N}$  concentration. The results suggested that LiH and  $\text{Li}_3\text{N}$  played similar roles in  $\text{NH}_3$  absorption based on reactions (7) and (8). The flattened  $\text{NH}_3$  synthesis rate vs. time curves in Fig. 4g and h were enabled jointly by  $\text{Li}_3\text{N}$  and LiH in the molten salt. The results also indicated that controlling the concentrations of  $\text{Li}_3\text{N}$  and LiH in the molten salt was vital for improving the  $\text{NH}_3$  synthesis rate.

**3.2.3. HCl as an alternative proton source to react with  $\text{Li}_3\text{N}$  in the molten salt.** Based on the experimental results above, the reaction of  $\text{H}_2$  with  $\text{Li}_3\text{N}$  in the molten salt led to the accumulation of relatively stable intermediate species (e.g.,  $\text{NH}^{2-}$ ,  $\text{NH}_2^-$ , and  $\text{H}^-$ ). Although the  $\text{NH}^{2-}/\text{NH}_2^-$  species could react with  $\text{H}_2$  to produce  $\text{NH}_3$  (reactions (4b) and (4c); Fig. S1†), these reactions were thermodynamically unfavorable (Fig. S1b†). Therefore, the relatively stable  $\text{NH}^{2-}/\text{NH}_2^-$  intermediates resulted in a significant amount of N-containing species remaining in the molten salt, instead of all converting to  $\text{NH}_3$ . In other words, the accumulation of the  $\text{NH}^{2-}/\text{NH}_2^-$  species in the molten salt corresponded to the limited  $\text{NH}_3$  synthesis. Here, we investigated the use of hydrogen chloride (HCl) gas as an alternative proton source to react with  $\text{Li}_3\text{N}$  in the molten salt to produce  $\text{NH}_3$ , as expressed by reaction (9).<sup>63</sup> We performed two control experiments of feeding 25%  $\text{H}_2/\text{Ar}$  (20  $\text{cm}^3 \text{min}^{-1}$ ) and 50%  $\text{HCl}/\text{Ar}$  (20  $\text{cm}^3 \text{min}^{-1}$ ), respectively, into LiCl–KCl melts containing 0.5 mol%  $\text{Li}_3\text{N}$  for 1 hour at 500 °C. Afterwards, pure Ar (20  $\text{cm}^3 \text{min}^{-1}$ ) was bubbled to the melt for another 3 hours to purge  $\text{NH}_3$  before cooling down the reactor in each experiment.  $\text{NH}_3$  was collected throughout the 4 hours using the 0.05  $\text{mol L}^{-1}$  aqueous  $\text{H}_2\text{SO}_4$  solution. It's noted that dry 50%  $\text{HCl}/\text{Ar}$  (20  $\text{cm}^3 \text{min}^{-1}$ ) was made in the lab by passing a mixture of  $\text{H}_2$  (5  $\text{cm}^3 \text{min}^{-1}$ ) and Ar (10  $\text{cm}^3 \text{min}^{-1}$ ) through a quartz tube containing anhydrous  $\text{NiCl}_2$  at 700 °C in a horizontal furnace. Fig. 4i shows the nitrogen weight fractions in the solidified salts and the total amounts of ammonia collected in the  $\text{H}_2\text{SO}_4$  solutions. A comparison of the results indicated that HCl was much more effective than  $\text{H}_2$  in converting  $\text{Li}_3\text{N}$  to  $\text{NH}_3$ . In the reaction between  $\text{H}_2$  and  $\text{Li}_3\text{N}$  in the molten salt, HCl might be introduced to the melt occasionally to control the nitrogen concentration and improve the  $\text{NH}_3$  synthesis rate.



In addition to  $\text{NH}_3$ , reaction (9) also produces LiCl which in principle can be electrolyzed to generate Li and  $\text{Cl}_2$ . Li can be used to fix  $\text{N}_2$  and form  $\text{Li}_3\text{N}$ , while  $\text{Cl}_2$  can react with  $\text{H}_2$  to regenerate HCl, which would react with  $\text{Li}_3\text{N}$  to produce more  $\text{NH}_3$  according to reaction (9). This closed-loop process for improving  $\text{NH}_3$  synthesis is beyond the scope of the current study, and will be investigated in the future work.



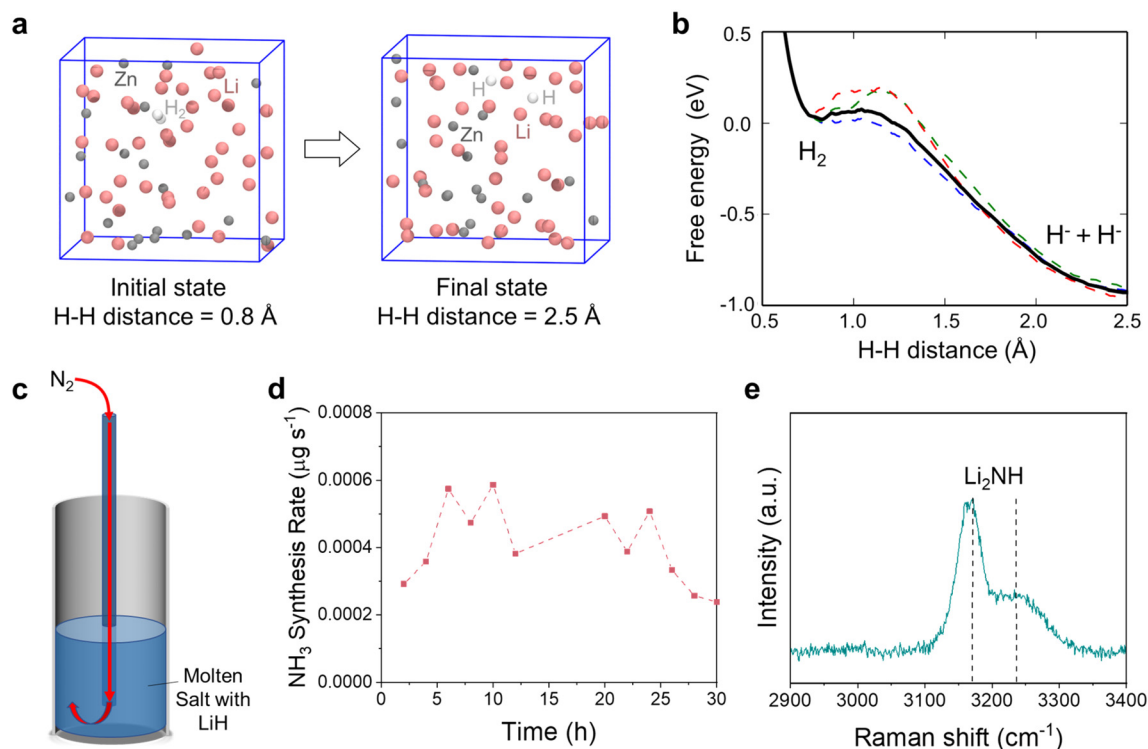
### 3.3. The reaction of LiH with N<sub>2</sub> for NH<sub>3</sub> synthesis in the molten salt

LiH is a byproduct of the reaction between H<sub>2</sub> and Li<sub>3</sub>N in the molten salt based on reaction (4). The accumulation of LiH in the molten salt has been identified to reduce the thermodynamic driving force for NH<sub>3</sub> synthesis.<sup>40</sup> The solid LiH is known to have a high thermodynamic stability due to the presence of a strong ionic bond, and its decomposition to Li and H<sub>2</sub> typically requires high temperatures above 700 °C.<sup>81</sup> To destabilize the LiH dissolved in the molten salt, previous work has proposed using a third element (*e.g.*, Sn) to alloy with Li upon LiH decomposition. This method of alloying lowered the energy barrier for LiH decomposition; however, LiH decomposition was still a thermodynamically unspontaneous reaction at the operation temperatures. The effectiveness of Li recovery from LiH decomposition with this method remains to be critically determined.

To evaluate the difficulty of LiH decomposition in the presence of the Li–Zn alloy (reaction (5)), we used AIMD to simulate the reverse reaction and investigate the reactivity of H<sub>2</sub> with the 70 mol% Li–Zn alloy. A randomly mixed Li<sub>42</sub>Zn<sub>18</sub> system was built in a periodic cubic cell with the edge length = 11.2 Å, and a H<sub>2</sub> molecule was placed inside, as shown in Fig. 5a. The AIMD simulation was performed at 400 °C.

Fig. 5b shows the free energy change for H<sub>2</sub> cleavage to two H<sup>−</sup> as a function of the H–H distance. The  $G_a$  for H<sub>2</sub> cleavage in the liquid Li–Zn alloy was merely 0.1 eV, much smaller than that in the gas phase. On the other hand, the  $G_a$  for LiH decomposition to form H<sub>2</sub> and Li–Zn alloy was identified to be merely 1 eV, validating reaction (5). Nevertheless, the results verified that LiH decomposition was unspontaneous; even with the presence of the Li–Zn alloy, the energy barrier of its decomposition was still much greater than that of its reverse reaction.

Given the difficulty of LiH thermal decomposition, here we investigated the reaction of LiH in the molten salt with N<sub>2</sub>. The eutectic LiCl–KCl mixture containing 3 mol% LiH was first heated to 400 °C in Ar, and then fed with N<sub>2</sub> (15 cm<sup>3</sup> min<sup>−1</sup>) to the bottom of the molten salt (Fig. 5c). Fig. 5d shows that the NH<sub>3</sub> synthesis rate was relatively small over 30 hours at 400 °C, indicating that the reaction of LiH with N<sub>2</sub> contributed little to NH<sub>3</sub> synthesis. Nevertheless, the solidified salt contained approximately 0.05 wt% of nitrogen based on the UV-Vis characterization (Fig. S15<sup>†</sup>), almost three times greater than that in the commercial LiCl–KCl salt (Fig. S7<sup>†</sup>), indicating nitrogen fixation by LiH. Raman spectroscopy of the solidified salt shows characteristic peaks at 3170 cm<sup>−1</sup> and 3230 cm<sup>−1</sup> for Li<sub>2</sub>NH (Fig. 5e),<sup>77</sup> indicating that the major product of the reaction between LiH and N<sub>2</sub>

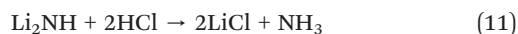
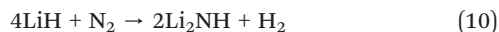


**Fig. 5** (a) Snapshots of the initial state and the final state of the model system where the cubic box represents the periodic unit cell of the infinite bulk system. Color code: pink, Li; gray, Zn; white, H. (b) The free energy profile as a function of the H–H distance during the H<sub>2</sub> cleavage process, as denoted by the black bold line. The work profiles of 3 slow-growth trajectories are also shown, as denoted by the dashed lines in color. (c) Schematic diagram of the experimental setup for bubbling N<sub>2</sub> (5 cm<sup>3</sup> min<sup>−1</sup>) to the eutectic LiCl–KCl molten salt (height: 5 cm) containing 3 mol% LiH. (d) The NH<sub>3</sub> synthesis rate measured for the experiment illustrated in (c) over 30 hours at 400 °C. (e) Raman spectrum of the solidified salt sample from the experiment showing characteristic peaks for Li<sub>2</sub>NH.





was  $\text{Li}_2\text{NH}$  (reaction (10)). The degree of nitridation of  $\text{LiH}$  dissolved in the molten  $\text{LiCl-KCl}$  salt was calculated to be 13.18% at 400 °C (Note S5†). The dissolution of  $\text{LiH}$  in the molten salt enabled a gas-liquid reaction between  $\text{LiH}$  and  $\text{N}_2$  in the bubbles. Unlike the gas-solid reaction between  $\text{N}_2$  and solid  $\text{LiH}$ , the  $\text{Li}_2\text{NH}$  product dissolved into the molten salt without forming a blocking layer to hinder further reaction. In addition,  $\text{Li}_2\text{NH}$  may be further utilized by reacting with  $\text{H}_2$  to form  $\text{NH}_3$  (reactions (4b) and (4c)), but the resulting  $\text{NH}_3$  synthesis rate was supposed to be limited as these reactions were thermodynamically unfavorable (Fig. S1b†). In principle,  $\text{Li}_2\text{NH}$  in the molten salt may also be converted to  $\text{NH}_3$  by using  $\text{HCl}$  gas as an alternative and more effective proton source (reaction (11)), as discussed in section 3.2.3.



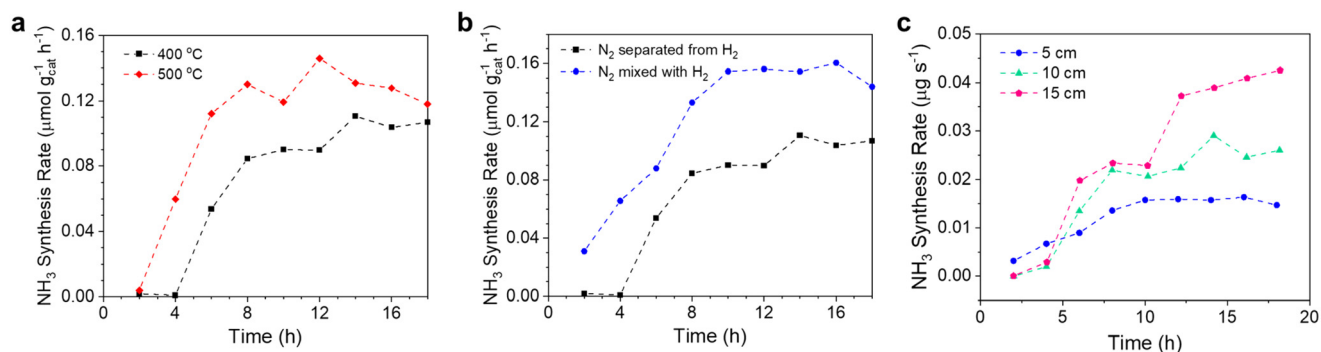
### 3.4. $\text{NH}_3$ synthesis in the liquid alloy-salt catalytic system

The combined experiments and simulations above have shown that the  $\text{Li-Zn}$  alloy is effective for fixing  $\text{N}_2$ , and the hydrogenation of  $\text{Li}_3\text{N}$  dissolved in the molten salt is reactive. We then investigated the effects of multiple variables on  $\text{NH}_3$  synthesis from  $\text{N}_2$  and  $\text{H}_2$  in the molten catalytic system comprising the  $\text{Li-Zn}$  alloy and the  $\text{LiCl-KCl}$  salt.

**3.4.1. The effect of temperature.** We performed two experiments with the catalytic systems comprising 70 mol%  $\text{Li-Zn}$  alloy (height: 5 cm) and  $\text{LiCl-KCl}$  salt (height: 5 cm) at 400 °C and 500 °C, respectively.  $\text{N}_2$  ( $5 \text{ cm}^3 \text{ min}^{-1}$ ) and  $\text{H}_2$  ( $15 \text{ cm}^3 \text{ min}^{-1}$ ) were fed to the liquid alloy and molten salt, respectively (Fig. 1a). The  $\text{NH}_3$  synthesis rates were measured over 18 hours at these temperatures; the data exhibited similar trends, starting with a gradual increase and then

reaching steady values (Fig. 6a). Notably, in the first 4 hours, the  $\text{NH}_3$  synthesis rate at 400 °C was almost zero, whereas the  $\text{NH}_3$  synthesis rate at 500 °C was relatively large. The stable value of the  $\text{NH}_3$  synthesis rate at 500 °C was also slightly greater than that at 400 °C, albeit with some fluctuations. This difference was attributed to the following factors. First,  $\text{NH}_3$  synthesis from  $\text{N}_2$  and  $\text{H}_2$  is an exothermic and entropy-decreasing reaction. Although lower temperature was thermodynamically beneficial to  $\text{NH}_3$  yield, this reaction was kinetically restricted due to the stable  $\text{N}\equiv\text{N}$  bonds and the short residence time of the bubbles in the alloy column (5 cm).<sup>5</sup> Increasing the temperature generally promoted the kinetics of nitrogen fixation. Second, the  $\Delta G^\circ$  of the reaction between  $\text{Li}_3\text{N}$  and  $\text{NH}_3$  (reaction (7)) became less negative with the increase of temperature (Fig. S1b†), weakening the  $\text{NH}_3$  absorption and increasing the  $\text{NH}_3$  release. As shown in Table S4†, the N weight fraction in the solidified  $\text{LiCl-KCl}$  salt from the experiment at 500 °C was 0.783 wt%, still larger than that at 400 °C (0.559 wt%). The results indicated that, although the  $\text{NH}_3$  absorption was weakened, the nitrogen fixation ability of Li was stronger at a higher temperature. Nevertheless, it should be noted that the steady values for  $\text{NH}_3$  synthesis at 400 °C and 500 °C were relatively close, indicating it suitable to operate at relatively low temperatures, especially considering the development trend toward milder and greener conditions for  $\text{NH}_3$  synthesis.

**3.4.2. The effect of the feeding mode of  $\text{N}_2$  and  $\text{H}_2$ .** There were different ways of feeding  $\text{N}_2$  ( $5 \text{ cm}^3 \text{ min}^{-1}$ ) and  $\text{H}_2$  ( $15 \text{ cm}^3 \text{ min}^{-1}$ ) into the molten catalytic system comprising 70 mol%  $\text{Li-Zn}$  (height: 5 cm) and  $\text{LiCl-KCl}$  (height: 5 cm). In the first way of feeding  $\text{N}_2$  and  $\text{H}_2$  separately into the liquid alloy and the molten salt, the  $\text{NH}_3$  synthesis rate gradually increased with time and stabilized at  $\sim 0.11 \mu\text{mol g}_{\text{cat}}^{-1} \text{ h}^{-1}$  (equivalent to  $\sim 0.012 \mu\text{g s}^{-1}$ ) (Fig. 6b). Meanwhile, in the second way of feeding the  $\text{N}_2$  and  $\text{H}_2$  mixture into the bottom of the liquid alloy, the  $\text{NH}_3$  synthesis rate followed a similar trend and stabilized at  $\sim 0.15 \mu\text{mol g}_{\text{cat}}^{-1} \text{ h}^{-1}$  (equivalent to



**Fig. 6** (a) The  $\text{NH}_3$  synthesis rate in liquid catalytic systems comprising  $\text{Li-Zn}$  alloy (height: 5 cm) and molten  $\text{LiCl-KCl}$  (height: 5 cm) at 400 and 500 °C.  $\text{N}_2$  ( $5 \text{ cm}^3 \text{ min}^{-1}$ ) and  $\text{H}_2$  ( $15 \text{ cm}^3 \text{ min}^{-1}$ ) were fed to the alloy and the salt, respectively. (b) The  $\text{NH}_3$  synthesis rate in liquid catalytic systems comprising  $\text{Li-Zn}$  alloy (height: 5 cm) and molten  $\text{LiCl-KCl}$  (height: 5 cm) at 400 °C. In one experiment as denoted by black squares,  $\text{N}_2$  ( $5 \text{ cm}^3 \text{ min}^{-1}$ ) and  $\text{H}_2$  ( $15 \text{ cm}^3 \text{ min}^{-1}$ ) were fed to the alloy and salt, respectively. In the other experiment as denoted by the blue spheres, the mixture of  $\text{N}_2$  ( $5 \text{ cm}^3 \text{ min}^{-1}$ ) and  $\text{H}_2$  ( $15 \text{ cm}^3 \text{ min}^{-1}$ ) was fed to the liquid alloy. (c) The  $\text{NH}_3$  synthesis rate in liquid catalytic systems comprising  $\text{Li-Zn}$  (5 cm) and  $\text{LiCl-KCl}$  of different heights (5, 10, and 15 cm). The mixture of  $\text{N}_2$  ( $5 \text{ cm}^3 \text{ min}^{-1}$ ) and  $\text{H}_2$  ( $15 \text{ cm}^3 \text{ min}^{-1}$ ) was directly introduced to the liquid alloy.



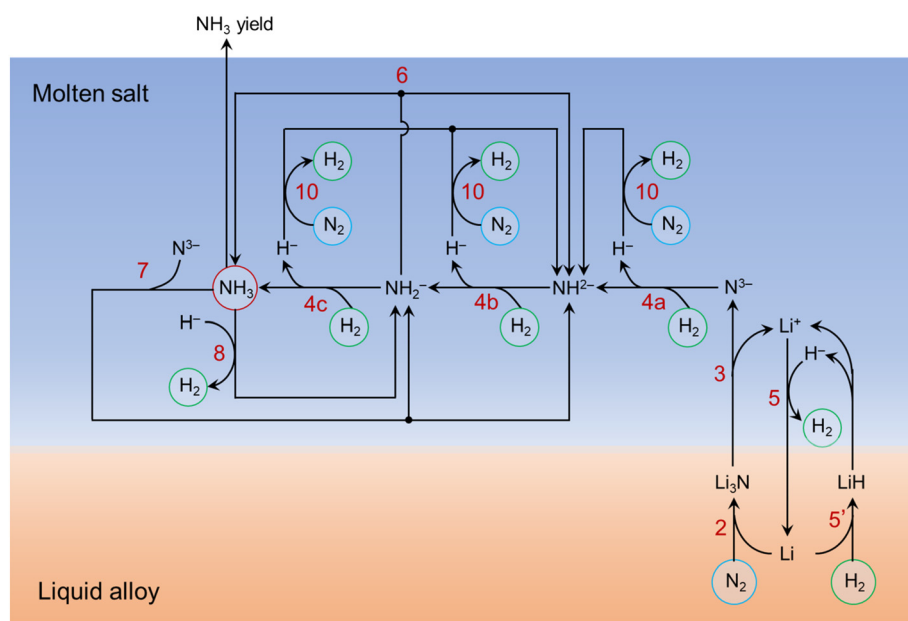
$\sim 0.016 \mu\text{g s}^{-1}$ ) (Fig. 6b). The improvement in the  $\text{NH}_3$  synthesis rate was enabled by changing the way of flowing  $\text{N}_2$  and  $\text{H}_2$  into the catalytic system. Such improvement was likely attributed to the following factors. First, in the case of mixing  $\text{N}_2$  and  $\text{H}_2$ , the partial pressure of  $\text{N}_2$  reacting with the 70 mol% Li–Zn alloy was lower, so the nitrogen fixation rate and the amount of  $\text{Li}_3\text{N}$  formed decreased, resulting in a weakened absorption of  $\text{NH}_3$ . The effect of  $\text{N}_2$  partial pressure on the nitrogen fixation rate was further verified by performing control experiments (see more details in Note S6†). Second, when the mixture of  $\text{N}_2$  and  $\text{H}_2$  was passed into the bottom of the Li–Zn alloy, the reaction ( $2\text{Li} + \text{H}_2 \rightarrow 2\text{LiH}$ ) also consumed a portion of Li, which competed with the nitrogen fixation and lowered the amount of  $\text{Li}_3\text{N}$  produced.<sup>45</sup> In the case of feeding the gas mixture, the N weight fraction in the solidified salt was indeed lower (0.407 wt%) (Table S4†). Third,  $\text{H}_2$  can react with  $\text{Li}_3\text{N}$  the moment it was formed in the alloy when feeding the  $\text{N}_2$ – $\text{H}_2$  mixture, increasing the reaction time and enabling more reaction pathways for  $\text{NH}_3$  synthesis. Therefore, the  $\text{NH}_3$  synthesis rate was greater in the case of feeding the mixture of  $\text{N}_2$  and  $\text{H}_2$  than feeding them separately. Feeding the mixture of  $\text{N}_2$  and  $\text{H}_2$  reduces the complexity of the setup, and brings the system closer to industrial reaction conditions where the unreacted  $\text{N}_2$  and  $\text{H}_2$  mixture can be recycled back into the reactor after ammonia separation.<sup>10</sup>

**3.4.3. The effect of the molten salt height.** The effect of the height of the eutectic LiCl–KCl molten salt on the  $\text{NH}_3$  synthesis rate was also investigated. We performed three experiments on the molten catalytic systems comprising 70 mol% Li–Zn of a fixed height of 5 cm and LiCl–KCl of three different heights (5, 10, and 15 cm) at 400 °C. The mixture of

$\text{N}_2$  ( $5 \text{ cm}^3 \text{ min}^{-1}$ ) and  $\text{H}_2$  ( $15 \text{ cm}^3 \text{ min}^{-1}$ ) was fed into the bottom of the liquid alloy. Fig. 6c shows the  $\text{NH}_3$  synthesis rate monitored over 18 hours in these experiments, which exhibited similar trends, starting with a gradual increase in the first 10 hours and then reaching relatively steady values. As the height of the molten salt column increased from 5 cm, 10 cm, to 15 cm, the  $\text{NH}_3$  synthesis rate at the 18th hour increased from  $0.015 \mu\text{g s}^{-1}$  ( $0.121 \mu\text{mol g}_{\text{cat}}^{-1} \text{ h}^{-1}$ ),  $0.026 \mu\text{g s}^{-1}$  ( $0.132 \mu\text{mol g}_{\text{cat}}^{-1} \text{ h}^{-1}$ ), to  $0.043 \mu\text{g s}^{-1}$  ( $0.137 \mu\text{mol g}_{\text{cat}}^{-1} \text{ h}^{-1}$ ), respectively. After cooling to room temperature, the solidified salt samples were characterized to measure the nitrogen content. The nitrogen weight fraction decreased from 0.407 wt%, 0.304 wt%, to 0.276 wt%, with the increase of the salt column height (Table S4†). The decrease of the  $\text{Li}_3\text{N}$  concentration in this range weakened the  $\text{NH}_3$  absorption and thereby increased the  $\text{NH}_3$  synthesis rate, based on the volcano plot (Fig. 4d). The results indicate the feasibility of raising the height of molten LiCl–KCl salt for improving the  $\text{NH}_3$  synthesis rate.

### 3.5. Summary of major reactions in the liquid catalytic system and the outlook

Improving the understanding of the molten catalytic system provides a guideline for designing better molten catalytic systems. The lithium-based loop presented in the previous work is over-simplified.<sup>40</sup> Based on the afore-mentioned experimental and simulation results, here we have proposed a more complete picture of the major reactions taking place in the molten catalytic system for  $\text{NH}_3$  synthesis from  $\text{N}_2$  and  $\text{H}_2$ , as illustrated in Fig. 7. The ammonia synthesis process involves multiple reaction steps especially in the molten salt.



**Fig. 7** A holistic illustration of the major reactions taking place in the molten catalytic system comprising Li–Zn alloy and LiCl–KCl salt. The gaseous molecules ( $\text{N}_2$ ,  $\text{H}_2$ ,  $\text{NH}_3$ ) are circled to increase clarity. The numbers in red refer to the reactions listed in the main text. The effect of HCl is not considered here. Reaction (5') is the reverse of reaction (5).



It has taken into account the stepwise reactions of  $\text{N}^{3-}$  hydrogenation (reactions (4a)–(4c)), the absorption of  $\text{NH}_3$  by  $\text{N}^{3-}$  (reaction (7)) and  $\text{H}^-$  (reaction (8)), and  $\text{N}_2$  fixation by  $\text{H}^-$  (reaction (10)). Reactions (6), (7), (8), and (10) lead to the accumulation of the relatively stable intermediate species (*i.e.*,  $\text{NH}^{2-}$ ,  $\text{NH}_2^-$ ) in the molten salt, corresponding to the limited  $\text{NH}_3$  yield, as discussed in section 3.2.3.

To evaluate the catalytic performance of the molten catalytic system for ammonia synthesis from  $\text{N}_2$  and  $\text{H}_2$ , we have compared the activity levels of the molten catalytic systems studied in this work with several relevant catalysts reported in the literature, as summarized in Table S5.† The highest activity level obtained was  $0.137 \mu\text{mol g}_{\text{cat}}^{-1} \text{h}^{-1}$  in the molten catalytic system comprising 70 mol% Li–Zn (5 cm) and eutectic LiCl–KCl (15 cm) at 400 °C and 1 bar. This value was greater than that obtained in pure liquid sodium<sup>29</sup> but smaller than those of solid catalysts such as the Zn/LiH system,<sup>59</sup> La–TM–Si systems (TM=Co, Fe, Mn),<sup>28</sup> and Cs-promoted Ru/MgO.<sup>82</sup> Overall, the molten catalytic system has a significant ammonia yield rate but a low calculated activity level. The reason is actually clear. The molten catalytic system uses a heavy bubble column (21.55–39.15 g) with large-size bubbles at a limited flow rate ( $5\text{--}20 \text{ cm}^3 \text{ min}^{-1}$ ), and the effective gas/liquid interfacial areas for nitrogen fixation and ammonia synthesis are small; in comparison, the solid catalysts reported in the literature<sup>28,59,82</sup> use much less material (0.03–0.1 g) with large surface areas and high gas flows.

Apparently, there are opportunities for improving the molten catalytic system for ammonia synthesis significantly. For example, the material selection for the liquid alloy and molten salt is not restricted to Li–Zn and LiCl–KCl. It is necessary to investigate other reactive alloys and ionic salts. It is also critical to advance the understanding of the high atomic mobility and the complex atomic structure in the melts of various compositions with *in situ* characterization techniques<sup>83</sup> and artificial intelligence methods,<sup>84</sup> which might considerably improve the catalytic performance of the system. The dynamic interface between the alloy and the salt is also of great interest. In addition, it will be necessary to carry out scale-up studies and develop pressurized reactors that allow the generation of a significant amount of very tiny gas bubbles to increase the area of the gas/liquid reaction interface. Besides, as indicated by Fig. 4i, the introduction of a suitable amount of HCl into the molten salt can improve the ammonia synthesis rate significantly due to rapid reactions of HCl with  $\text{N}^{3-}$ ,  $\text{NH}^{2-}$ , and  $\text{NH}_2^-$ . HCl readily disturbs the accumulation of the relatively stable  $\text{NH}^{2-}$  and  $\text{NH}_2^-$  in the molten salt. Moreover, the liquid alloy–salt catalytic system can incorporate electrolysis to facilitate the ammonia synthesis and remove the oxide impurities if formed by the reaction between the reactive metal and any water or oxygen present in the feeding gas. These exciting research opportunities are worthy of thorough investigations in the future work.

## 4. Conclusions

The liquid alloy–salt catalytic system is an integrated, synergistic, and multi-phase (*i.e.*, liquid alloy, molten salt, gas bubbles) system that offers unique chemistry and reaction pathways for ammonia synthesis under relatively mild conditions. Through combined experiments and simulations, we have studied the mechanism of nitrogen fixation and ammonia synthesis in the catalytic system comprising liquid Li–Zn and molten LiCl–KCl. The 70 mol% Li–Zn liquid alloy is effective for  $\text{N}_2$  dissociation and fixation, and Li is identified to be the reactive species.  $\text{Li}_3\text{N}$  floats from the self-cleaning liquid alloy to the highly ionic molten salt, circumventing the scaling relation exerted on Li. When feeding  $\text{H}_2$  to the molten salt, the relationship between the  $\text{Li}_3\text{N}$  concentration in the molten salt and the  $\text{NH}_3$  synthesis rate exhibits a volcano plot. In the liquid catalytic system,  $\text{Li}_3\text{N}$  is first formed in the alloy without LiH. The  $\text{NH}_3$  generated from  $\text{Li}_3\text{N}$  and  $\text{H}_2$  in the molten salt might be absorbed by the excessive  $\text{Li}_3\text{N}$ , limiting the  $\text{NH}_3$  synthesis rate. As the reactions proceed, LiH accumulates and absorbs  $\text{NH}_3$  as well. These findings provide guidelines for the optimization of the experiments. Different strategies have been investigated for improving the  $\text{NH}_3$  synthesis rate, including (i) feeding HCl to the salt, (ii) changing the temperature, (iii) flowing the mixture of  $\text{N}_2$  and  $\text{H}_2$  to the alloy, and (iv) increasing the height of the molten salt column. In summary, this work has demonstrated the effectiveness of the catalytic system based on liquid Li–Zn and molten LiCl–KCl for  $\text{NH}_3$  synthesis at relatively low temperatures (400–500 °C) and ambient pressure, and will motivate further research and development on decentralized ammonia synthesis with complex melts.

## Author contributions

X. G. conceived the study and supervised the project. X. M., Z. T., and B. X. performed the experiments, characterized the materials, and analyzed the data under X. G.'s guidance. J. L. and B. Y. designed and performed the AIMD simulations, and analyzed the computational results. P. Y. performed the STEM and EELS experiments and analyzed the results under K. C.'s guidance. X. W. assisted in the material characterization and the technical discussion. X. M., J. L., Z. T., B. X., B. Y., and X. G. wrote the manuscript with feedback from all authors.

## Conflicts of interest

The following may be considered as potential conflict of interest: X. G., Z. T., and X. M. have patents (CN112266002A, CN112250088A) issued to ShanghaiTech University.

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