



Cite this: *Phys. Chem. Chem. Phys.*,  
2024, 26, 22454

## Anisotropic atomic displacements, local orthorhombicity and anomalous local magnetic moment in $\text{Ba}_{0.6}\text{K}_{0.4}\text{Fe}_2\text{As}_2$ superconductor

L. Tortora,<sup>a</sup> G. Tomassucci,<sup>ID a</sup> G. M. Pugliese,<sup>a</sup> M. Y. Hacisalihoglu,<sup>ID a</sup> L. Simonelli,<sup>ID b</sup> C. Marini,<sup>b</sup> G. Das,<sup>ID c</sup> S. Ishida,<sup>ID d</sup> A. Iyo,<sup>ID d</sup> H. Eisaki,<sup>ID d</sup> T. Mizokawa,<sup>ID e</sup> and N. L. Saini,<sup>ID \*a</sup>

We have investigated the in-plane local structure of the  $\text{Ba}_{0.6}\text{K}_{0.4}\text{Fe}_2\text{As}_2$  superconductor by polarized Fe K-edge extended X-ray absorption fine structure (EXAFS) measurements with temperature. The near neighbor bond distances and their stiffness, measured by polarized EXAFS in two orthogonal directions, are different suggesting in-plane anisotropy of the atomic displacements and local orthorhombicity in the title system. The X-ray absorption near edge structure (XANES) spectra reveal anisotropy of valence electronic structure that changes anomalously below  $\sim 100$  K. The local iron magnetic moment, measured by Fe K $\beta$  X-ray emission spectroscopy (XES), increases below the anomalous temperature and shows a decrease in the vicinity of the superconducting transition temperature ( $T_c \sim 36$  K). The results provide a clear evidence of coupled local lattice, electronic and magnetic degrees of freedom to induce possible nematic fluctuations in an optimally hole doped iron-based superconductor.

Received 9th June 2024,  
Accepted 2nd August 2024

DOI: 10.1039/d4cp02345e

rsc.li/pccp

### 1 Introduction

Iron-based superconductors have been the focus of debate and, in particular, for the still unknown nature of electron pairing in these materials. These systems are in fact characterized by a complex temperature-doping phase diagram, in which different structural and magnetic phases are often overlapped.<sup>1–5</sup> Almost all members of this family present a tetragonal unit cell at room temperature, which undergoes a structural (tetragonal to orthorhombic) and magnetic (to striped antiferromagnetic (AFM)) phase transition at lower temperature. These transitions are suppressed by doping with the emergence of a superconducting dome characterized by a  $T_c$ , up to  $\sim 55$  K. In addition, these materials are characterized by a multiband electronic structure in which five Fe 3d bands are filled by six electrons resulting in a complex orbital structure to form the Fermi surface.<sup>6,7</sup> The presence of Fermi surface nesting along the  $(0, \pi)$  or  $(\pi, 0)$  directions led many to speculate that some sort of

spin-density-wave (SDW) related pairing might be driving the superconductivity in these compounds, although the idea remains heavily debated.<sup>8,9</sup> The multi-orbital electronic state makes it hard to identify the leading order parameter for the phase transitions in iron-based compounds, as the magnetic, orbital and structural degrees of freedom of these systems are intertwined.<sup>5,10,11</sup> Furthermore, exotic magnetic phases have been observed in some of these systems, such as a very rare hedgehog spin-vortex crystal phase in  $\text{CaK}(\text{Fe}_{1-x}\text{Ni}_x)_4\text{As}_4$  and  $\text{LaFeAs}_{1-x}\text{OP}_x$ ,<sup>12,13</sup> while a nematic-superconducting state is commonly found in iron-based superconductors.<sup>11,14–19</sup>

Usually, the insurgence of the nematic phase is accompanied by a symmetry breaking of the crystal unit cell, which loses the  $C_4$  rotational symmetry transition from the high temperature tetragonal phase to the low temperature orthorhombic configuration.<sup>11,14,15,18,19</sup> In most compounds this structural phase transition occurs at a temperature  $T_S$  in the vicinity of the magnetic transition temperature  $T_N$ , possibly suggesting that the highly discussed nematic order may be a consequence of a change in the local structure.<sup>20,21</sup> On the other hand, the magnetic order itself breaks the  $C_4$  rotational symmetry, making these compounds a viable choice for investigating such phenomena.

Among iron-based materials,  $\text{BaFe}_2\text{As}_2$  has tetragonal structure at room temperature (Fig. 1) that undergoes structural phase transition at  $T_S$  and magnetic phase transition at  $T_N$  (with  $T_S \sim T_N$ ) and, when doped, exhibits superconductivity in

<sup>a</sup> Dipartimento di Fisica, Università di Roma "La Sapienza" – P. le Aldo Moro 2, 00185 Roma, Italy. E-mail: naurang.saini@roma1.infn.it

<sup>b</sup> CELLS - ALBA Synchrotron Radiation Facility, Carrer de la Llum 2-26, 08290, Cerdanyola del Vallès, Barcelona, Spain

<sup>c</sup> Elettra Sincrotrone Trieste, Strada Statale 14, Km 163.5, Basovizza, 34149 Trieste, Italy

<sup>d</sup> National Institute of Advanced Industrial Science and Technology (AIST), Tsukuba, Ibaraki 305-8568, Japan

<sup>e</sup> Department of Applied Physics, Waseda University, Tokyo 169-8555, Japan



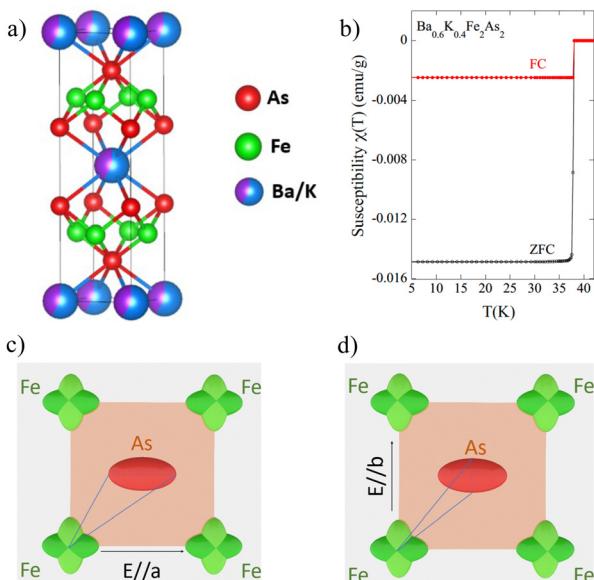


Fig. 1 (a) Crystal unit cell of  $(\text{Ba}, \text{K})\text{Fe}_2\text{As}_2$  system plotted by Vesta;<sup>32</sup> (b) temperature dependent magnetic susceptibility of  $\text{Ba}_{0.6}\text{K}_{0.4}\text{Fe}_2\text{As}_2$  single crystal showing a sharp superconducting transition at  $T_c \sim 36$  K; structural cartoons showing experimental geometries for the Fe K-edge, *i.e.*  $E//a$  (c) and  $E//b$  (d).

the proximity of the argued nematic quantum critical point. However, there is no evidence of any lower-symmetry crystal environment around the optimum doping although in-plane anisotropy has been indicated in different degrees of freedoms.<sup>22–29</sup> The in-plane anisotropy, assigned to nematic fluctuations, has been observed under uniaxial strain field also in doped iron-based compounds including in the hole doped  $\text{BaFe}_2\text{As}_2$ .<sup>30,31</sup>

On the other hand, local structure investigations by X-ray diffraction pair distribution function (PDF) on the isostructural  $\text{Sr}_{1-x}\text{Na}_x\text{Fe}_2\text{As}_2$  have revealed local orthorhombicity in the long range tetragonal phase,<sup>33,34</sup> assigned to the nematic fluctuations that may last up to much higher temperature than the  $T_s$ . The coherence length of the observed local orthorhombicity has been estimated to be in the range of 1–3 nm. Such a local orthorhombicity has been observed also in  $\text{NaFe}_{1-x}\text{Ni}_x\text{As}$  superconductor<sup>35</sup> using X-ray and neutron diffraction PDF. Also, similar approach has been used to unearth nematic fluctuations in a Mott insulator  $\text{La}_2\text{O}_2\text{Fe}_2\text{OM}_2$  ( $\text{M} = \text{S}, \text{Se}$ ).<sup>36,37</sup> In all these studies instantaneous local orthorhombic fluctuations slower than  $\sim 10^{-13}$  s have been probed.

In this work, we have exploited temperature dependent X-ray absorption fine structure (XAFS) spectroscopy,<sup>38–40</sup> a site selective local ( $\sim 1$  nm) and fast ( $\sim 10^{-15}$  s) experimental probe of instantaneous atomic displacements, to determine the local orthorhombicity in an optimally hole doped iron-based superconductor. In particular, we show that the local structure of  $\text{Ba}_{0.6}\text{K}_{0.4}\text{Fe}_2\text{As}_2$  is characterized by anisotropic atomic displacements and a local orthorhombicity. These displacements are probed by polarized extended X-ray absorption fine structure (EXAFS) analyses as a function of temperature on a single

crystal sample. In the EXAFS process, photoelectron excited from a core-level of the selected atom travels in the structure as an electron wave, that gets scattered from the near-neighbor atoms and hence generates oscillations in the energy dependent X-ray absorption coefficient. If the X-ray beam is polarized, the photoelectron wave travels in the polarization direction, therefore, the information contained is directional with the length scale limited by the mean-free path ( $\sim 1$  nm) of the excited photoelectron. In the present case, the polarization of the X-ray beam was confined within the Fe–Fe layer in the two orthogonal directions, *i.e.*  $E//a$  and  $E//b$  (see, *e.g.* the Fig. 1). Apart from differing bondlengths due to local orthorhombicity at a length scale of  $\sim 1$  nm, an anisotropic bond stiffness, determined by the temperature dependence of EXAFS Debye Waller factors<sup>38–40</sup> has been observed. The X-ray absorption near edge structure (XANES) spectra are consistent with the anisotropic local geometry and valence electronic structure. The valence electronic states show an anomalous change  $\sim 100$  K. The results on the local structure and the valence electronic states are combined with the study of the local magnetic moment of iron, determined by Fe K $\beta$  X-ray emission spectroscopy (XES)<sup>41</sup> that reveals an anomalous increase below  $\sim 100$  K followed by a decrease around the  $T_c \sim 36$  K of the system. The results are discussed in the frame of nematic fluctuations in the optimally doped iron-based superconductors.

## 2 Experimental details

A single crystal of  $\text{Ba}_{0.6}\text{K}_{0.4}\text{Fe}_2\text{As}_2$  sample showing a sharp superconducting transition below  $\sim 36.1$  K (Fig. 1b), was used for the XAFS and XES measurements. The sample, was characterized for its average structure and transport properties before the XAFS and XES measurements. Details on the synthesis and basic characterization of the sample can be found elsewhere.<sup>42</sup> The Fe K-edge (7112 eV) measurements were performed in transmission mode at the XAFS beamline<sup>43</sup> of Elettra synchrotron in Trieste (beam size  $\sim 500 \times 500 \mu\text{m}^2$ ). The bending magnet source at this beamline was monochromatized using a double crystal Si(111) monochromator with the measurement system containing three ionisation chambers: two for measuring the incident ( $I_0$ ) and transmitted ( $I_t$ ) X-ray intensity, and one for the transmission of a reference sample. The single crystal sample was exfoliated to obtain required thickness for the transmission measurements. The sample was rotated by  $90^\circ$  within the azimuth for the measurements to keep the polarization parallel to the  $a$  ( $E//a$ ) and  $b$  ( $E//b$ ) axes within the plane (Fig. 1). The measurements were repeated in part at the CLAES beamline<sup>44</sup> of the 3 GeV ALBA synchrotron radiation facility in Cerdanyola del Vallès (Barcelona) using similar transmission XAFS set-up (beam size  $\sim 200 \times 200 \mu\text{m}^2$ ).

The Fe K $\beta$  XES measurements were carried out using the CLEAR spectrometer<sup>45</sup> at the CLAES beamline, that is based on a diced Si(333) dynamically bent analyzer crystal and a position-sensitive Mythen detector. The data on the Fe K $\beta$  lines were acquired by exciting the sample well above the Fe K-edge,

with a total energy resolution of  $\sim 0.9$  eV. A minimum of four to five absorption/emission scans were acquired at each temperature. The sample temperature was carefully monitored and controlled with an accuracy of  $\pm 1$  K. A standard procedure based on spline fit was used for the EXAFS extraction<sup>38,39</sup> while the XANES spectra were normalized with respect to a linear fit to the data in the extended region after subtraction of pre-edge background.<sup>38</sup> Background signal was removed through the standard procedure performing a polynomial spline fit using the ATHENA software.<sup>46</sup>

### 3 Results and discussions

The Fe K-edge X-ray absorption fine structure measurements were carried out at different temperatures in the range of 20–300 K. Fig. 2 displays the Fourier transforms (FTs) of the EXAFS oscillations (multiplied by a factor  $k^2$ ) at several temperatures. The FTs are performed using a Gaussian window with the EXAFS  $k$ -range being  $2.8\text{--}15\text{ \AA}^{-1}$ . The upper panel corresponds to the data in the  $E\parallel a$  polarization (Fig. 1c) while the lower panel represents the  $E\parallel b$  polarization (Fig. 1d). The main peak in the Fe K-edge FT magnitude (left panel), centered at  $\sim 2.2\text{ \AA}$ , contains contribution of the nearest neighbor As atoms and the next nearest neighbor Fe atoms in the respective polarization directions. The longer distances contributions in the structure can be seen beyond  $\sim 3\text{ \AA}$  in the FTs of the EXAFS oscillations. The modulus of the FTs reveals some apparent differences in the atomic distribution in the two polarizations. One can make three observations, *i.e.*: (i) the overall magnitude of the main peak is lower in the  $E\parallel b$  direction; (ii) the position of the main FT peak differs slightly in the two polarizations; and (iii) the temperature dependence is different in the two polarizations. On the other hand, the longer distances contributions show highly damped magnitude indicating disordered structure of  $\text{Ba}_{0.6}\text{K}_{0.4}\text{Fe}_2\text{As}_2$ .

The differences in the local structure within the  $ab$ -plane can be quantified by EXAFS model fits. This has been done using the standard EXAFS equation with single scattering approximation.<sup>38,39</sup> The average structure measured by diffraction<sup>3,42</sup> has been the starting model for the Fe K-edge EXAFS containing the nearest neighbor As atoms and next nearest neighbor Fe atoms with two contributions mixed showing main peak as broad structure at  $\sim 2.2\text{ \AA}$ . The EXCURVE 9.275 code was used to calculate the backscattering amplitudes and the phase shift functions.<sup>47</sup> The same code was used to perform the EXAFS model fits, in which  $S_0^{-2}$  (amplitude reduction factor) and  $E_0$  (photoelectron energy zero) were fixed after a fit on the reference. The atomic positions  $R_i$  and related  $\sigma_i^{-2}$  (describing the Debye–Waller factor) were the only free parameters in the model fits. Within the reported uncertainties the same results were obtained using the scattering amplitudes and phase shifts calculated by FEFF code<sup>48</sup> and the model fits carried out by WINXAS package.<sup>49</sup> The number of independent data points,  $N_{\text{ind}} \sim \frac{2\Delta k \Delta R}{\pi}$ , was about 15 (interval in  $k$  space  $\Delta k = (15.0\text{--}2.8)\text{ \AA}^{-1}$  =  $12.2\text{ \AA}^{-1}$  and interval in  $R$  space  $\Delta R = (3.1\text{--}1.2)\text{ \AA} = 1.9\text{ \AA}$ ) for the

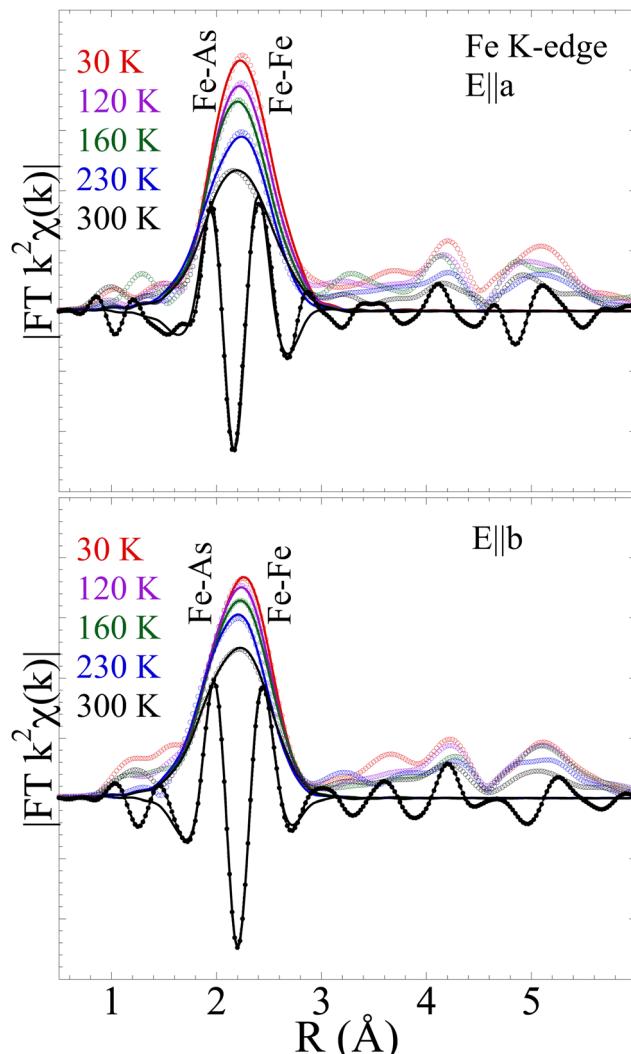
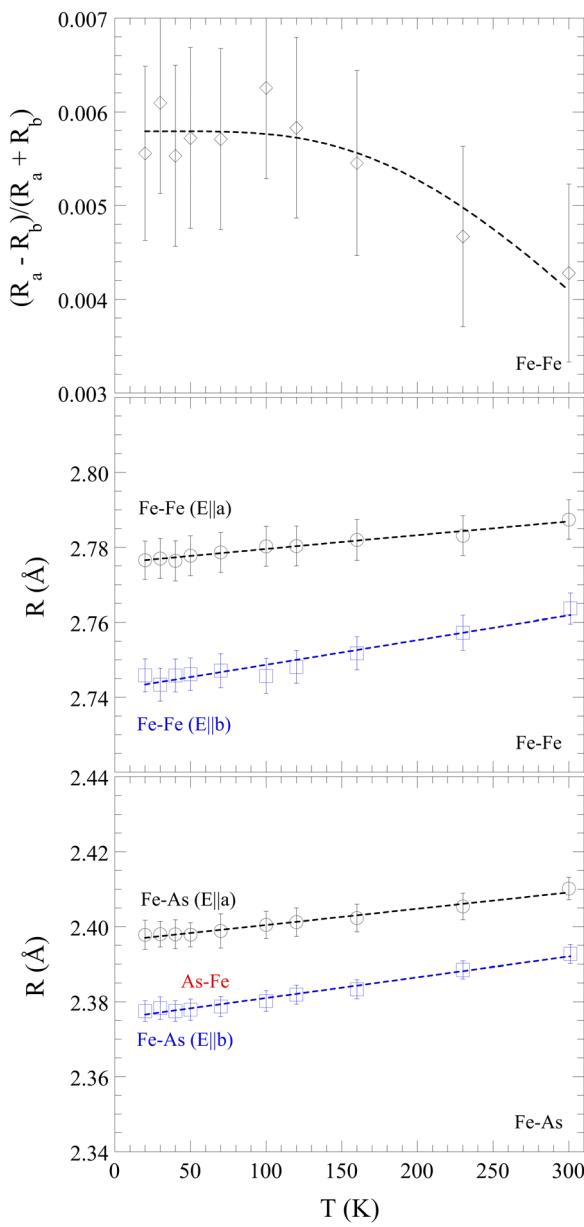


Fig. 2 Modulus of the Fourier transforms (FTs) of Fe K-edge EXAFS oscillations (multiplied by  $k^2$ ) measured on a single crystal sample of  $\text{Ba}_{0.6}\text{K}_{0.4}\text{Fe}_2\text{As}_2$  in  $E\parallel a$  (upper) and  $E\parallel b$  (lower) polarizations at various temperatures. The FTs are performed in the  $k$ -range of  $2.8$  and  $15\text{ \AA}^{-1}$  using a Gaussian window and not corrected by the phase shifts. The real parts of the FTs are also included to show differences in the two polarizations. The model fits in the real space (see text) are also included as solid lines.

Fe K-edge EXAFS models fits in which the number of variables were 4. The model fits in the real space are shown in Fig. 2 (solid lines).

Fig. 3 shows the near neighbors distances determined by the polarized EXAFS measured in the two orthogonal directions with the polarizations  $E\parallel a$  and  $E\parallel b$ . The local Fe–As bonds (lower panel), measured by Fe K-edge EXAFS in the  $E\parallel a$  direction is slightly longer ( $\sim 0.02\text{ \AA}$ ) than the same distance measured in the  $E\parallel b$  geometry. The bondlengths measured in the two geometries have similar thermal expansion and the difference between the two hardly shows any temperature dependence. The differing Fe–As bondlengths can be understood by simple geometrical considerations displayed pictorially in Fig. 1(c and d) in which the Fe atoms have symmetrical form in the  $ab$ -plane while the As atoms are asymmetric.





**Fig. 3** Near neighbor Fe-As (lower) and Fe-Fe (middle) bond lengths, determined by Fe K-edge EXAFS in the  $E\parallel a$  (black) and  $E\parallel b$  (blue) geometries are shown as a function of temperature for  $\text{Ba}_{0.6}\text{K}_{0.4}\text{Fe}_2\text{As}_2$ . The local orthorhombicity is shown as a function of temperature (upper). The dashed lines are to guide the eyes. The error bars show the maximum uncertainty determined by analysis of different EXAFS scans.

The asymmetrical form is related with differing occupation of  $p_x$  and  $p_y$  orbitals in the layered  $\text{Ba}_{0.6}\text{K}_{0.4}\text{Fe}_2\text{As}_2$ . Such an asymmetric shape will lead the Fe-As distance to be different in the two orthogonal polarizations if measured by the Fe K-edge. Indeed, linear dichroism in an isostructural  $\text{EuFe}_{1.8}\text{Co}_{0.2}\text{As}_2$  superconductor reveals uneven occupation of Fe  $d_{xz}$  and  $d_{yz}$  orbitals<sup>50</sup> hybridized with As  $4p_{x,y}$  orbitals.<sup>51</sup>

Similar to Fe-As, the Fe-Fe bondlengths are also different in the two orthogonal polarizations. The Fe-Fe distance, measured in the two polarizations can be different if there is a local

$C_4$  symmetry breaking. The local Fe-Fe bonds (middle panel), measured in the  $E\parallel a$  polarization is slightly longer ( $\sim 0.03$  Å) than the one measured in the  $E\parallel b$  geometry. Such a local orthorhombicity is consistent with the one measured in undoped  $\text{BaFe}_2\text{As}_2$ <sup>20,52</sup> at low temperature. The local orthorhombicity has been estimated as  $(R_a - R_b)/(R_a + R_b)$  and shown plotted in Fig. 3 (upper), where  $R_a$  and  $R_b$  are the Fe-Fe bond lengths measured in the two polarization geometries. Remarkably the temperature dependence of the local orthorhombicity is similar to the one found in  $\text{Sr}_{1-x}\text{Na}_x\text{Fe}_2\text{As}_2$  by PDF.<sup>33,34</sup> It should be noted that the local orthorhombicity measured in this work is slightly larger than the one measured by PDF, however, it is not surprising considering the fact that the EXAFS is more than hundred times faster than the PDF technique. Although small, the local orthorhombicity shows an increase at lower temperature. Nevertheless, the differing bondlengths in the two polarizations indicate local orthorhombicity and anisotropic atomic displacements within the  $ab$  plane. It should be mentioned that experimental approach used in this work is similar to the one already exploited for high  $T_c$  cuprates showing substantial differences in the bondlength distributions and the related atomic pair correlations.<sup>53,54</sup> Similar approach has been exploited recently to measure linear dichroism signal in an isostructural iron-based  $\text{EuFe}_{1.8}\text{Co}_{0.2}\text{As}_2$  superconductor revealing uneven occupation of Fe  $d_{xz}$  and  $d_{yz}$  orbitals.<sup>50</sup>

The differing temperature dependence of the FT amplitudes in Fig. 2 indicates different characteristics of the Fe-As and Fe-Fe bondlengths that can be quantified by the temperature dependent mean square relative displacement (MSRD) parameter  $\sigma_i^2$ . Indeed, bond stiffness is described by the Einstein frequency ( $\omega_E$ ), *i.e.*,  $\kappa = \mu\omega_E^2$  with the  $\mu$  being the reduced mass of the two bonded atoms. This information is accessible from the temperature dependence of  $\sigma_i^2$ . In the harmonic approximation the  $\sigma_i^2$  can be described by the Einstein model,<sup>39,40,55</sup> *i.e.*,

$$\sigma_i^2(T) = \sigma_0^2 + \frac{\hbar^2}{2\mu k_B \theta_E} \coth\left(\frac{\theta_E}{2T}\right) \quad (1)$$

where  $\theta_E$  is the Einstein temperature ( $\omega_E = k_B \theta_E / \hbar$ ). Here, the  $\sigma_0^2$  is the temperature independent part describing the configurational disorder, however, quantitative value of  $\sigma_0^2$  can be affected by the experimental conditions.

Fig. 4 shows the  $\sigma_i^2$  of Fe-As and Fe-Fe bondlengths measured by the Fe K-edge EXAFS in the two orthogonal directions. The temperature dependent behavior of the  $\sigma_i^2$  can be described by the Einstein model to determine the  $\theta_E$ , *i.e.* the Einstein frequency ( $\omega_E$ ). The Einstein temperatures determined for the Fe-As and Fe-Fe bondlengths are found to be  $\sim 339$  K and  $\sim 237$  K respectively in the  $E\parallel a$  polarization. The Einstein temperatures seem higher in the  $E\parallel b$  geometry, found to be  $\sim 365$  K and  $\sim 266$  K respectively for the two bondlengths suggesting anisotropic bond strength with the stiffer bonding along  $E\parallel b$ . The above results suggest anisotropic atomic fluctuations within the  $ab$ -plane of the optimally doped  $\text{Ba}_{0.6}\text{K}_{0.4}\text{Fe}_2\text{As}_2$  superconductor. Therefore, the broken



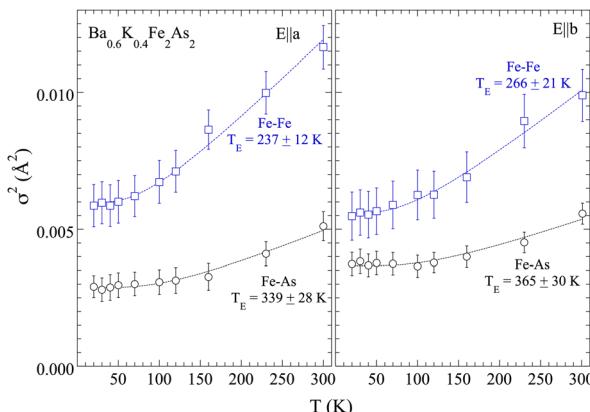


Fig. 4 Temperature dependence of the mean-square relative displacement parameter ( $\sigma^2$ ) of Fe-As (black) and Fe-Fe (blue) bondlengths measured by Fe K-edge EXAFS in the  $E \parallel a$  (left) and  $E \parallel b$  (right) polarizations on  $\text{Ba}_{0.6}\text{K}_{0.4}\text{Fe}_2\text{As}_2$ . The solid lines are the Einstein model fits (see text).

$C_4$  symmetry due to local orthorhombicity fluctuations, is characterized by anisotropic atomic displacements within the  $ab$ -plane. These details on the fluctuating nematic phase in the optimally doped  $\text{Ba}_{0.6}\text{K}_{0.4}\text{Fe}_2\text{As}_2$  superconductor are too subtle to be detected by long range experimental techniques without any external stimulant. Here, we are not ruling out mixing of nematic domains within the beam size with the observed anisotropy being due to inequivalent population of these domains. However, the fact that the anisotropy does not depend on the beam size, it should be intrinsic to the system.

The X-ray absorption near edge structure (XANES) is a very sensitive probe of the local geometry and the valence electronic states<sup>38,39</sup> that can add further information on the observed anisotropy. Fig. 5 shows normalized Fe K-edge XANES spectra of  $\text{Ba}_{0.6}\text{K}_{0.4}\text{Fe}_2\text{As}_2$  at 30 K (upper) and 300 K (lower) in the two orthogonal polarizations  $E \parallel a$  and  $E \parallel b$ . The XANES spectra are characterized by three main features  $A_1$ ,  $A_2$  and  $B$ , together with a clear pre-peak  $P$ . The features  $A_1$ ,  $A_2$  and  $B$ , well separated from the pre-peak  $P$ , are mainly due to the dipole allowed transitions from the Fe 1s to the unoccupied Fe 4p states in the continuum thus carrying useful information on the local geometry. The pre-peak feature  $P$  is due to quadrupole transition from the Fe 1s to the unoccupied Fe 3d states admixed with the p orbitals and directly probes the electronic structure of the unoccupied states and the Fe 3d-As 4p hybridization.<sup>51,56</sup>

The XANES spectra in the two polarizations reveal some apparent differences. For instance, the energy of the spectral features is higher in the  $E \parallel b$  geometry as shown by the spectral difference (multiplied by 2) displayed in Fig. 5. The maximum difference is  $\sim 6\%$  with a clear change in the pre-peak  $P$  as well as in the continuum part at energy less than that of the feature  $B$ . Indeed, the spectral weight of the pre-peak  $P$  and continuum features seems higher due to higher Fe 3d-As 4p hybridization and lower energy for the  $E \parallel a$  direction. The lower energy of XANES features in the  $E \parallel a$  polarization is consistent with the longer near neighbor distances observed by the EXAFS analysis. We have plotted the integrated intensity difference of the

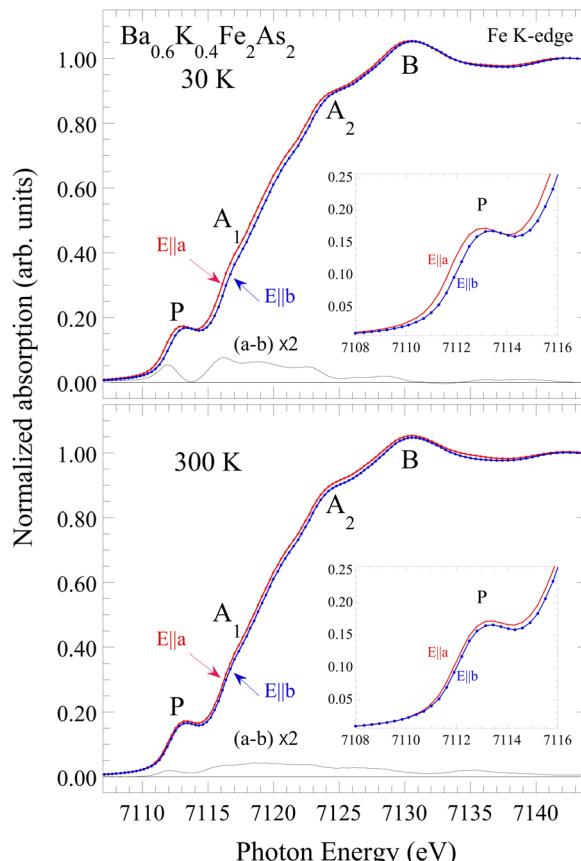


Fig. 5 Normalized Fe K-edge XANES spectra of  $\text{Ba}_{0.6}\text{K}_{0.4}\text{Fe}_2\text{As}_2$  at 30 K (upper) and at 300 K (lower) measured in two orthogonal polarizations within the  $ab$ -plane. The insets show zoom-overs the pre-peak  $P$  due to  $\text{Fe } 1s \rightarrow 3d$  transition. Direct spectral differences between the two polarizations (multiplied by 2) are also shown.

pre-edge peak  $P$  as a function of temperature in Fig. 6. The pre-peaks differ in their spectral weight even at 300 K due to differing hybridization, however this spectral difference shows

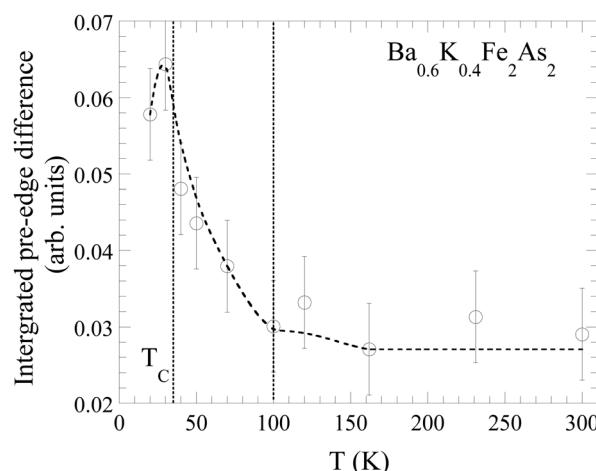


Fig. 6 Temperature dependence of the spectral difference of the Fe K-edge XANES pre-peak  $P$  for  $\text{Ba}_{0.6}\text{K}_{0.4}\text{Fe}_2\text{As}_2$ . Error bars represent the maximum uncertainty, and the dashed line serves only as a guide to the eyes.



a clear increase below  $\sim 100$  K. Since the pre-peak  $P$  is mainly given by the Fe 1s to 3d transition, the increased difference suggests increased difference in 3d–4p hybridization in the two directions below  $\sim 100$  K. Although present at higher temperature, the increased electronic anisotropy below  $\sim 100$  K is a signature of the electronic nematic fluctuations in  $\text{Ba}_{0.6}\text{K}_{0.4}\text{Fe}_2\text{As}_2$ .<sup>57–59</sup> It is also consistent with soft X-ray absorption study reporting uneven  $d_{xz}$  and  $d_{yz}$  population in  $\text{EuFe}_{1.8}\text{Co}_{0.2}\text{As}_2$ .<sup>50</sup>

It is fair to partially summarize the EXAFS and XANES results obtained on  $\text{Ba}_{0.6}\text{K}_{0.4}\text{Fe}_2\text{As}_2$ . Both EXAFS and XANES reveal a clear anisotropy in the first order correlation function (near neighbor distances) and in the higher order correlation function (local geometry). Since the length scale is  $\sim 1$  nm and the time scale is  $\sim 1$  fs, the anisotropy is due to instantaneous orthorhombic fluctuations in the system. The question is up to what extent the observed anisotropy can be associated with the commonly discussed nematic phase in the iron-based superconductors. The primary difficulty is due to the fact that the average orthorhombic phase with static antiferromagnetism (AFM) and orbital order in  $\text{BaFe}_2\text{As}_2$  is suppressed in the optimally hole doped  $\text{Ba}_{0.6}\text{K}_{0.4}\text{Fe}_2\text{As}_2$  even at low temperature, whereas the observed anisotropy is present at least up to 300 K. However, considering the fact that the intrinsic time scale of EXAFS and XANES is smaller than  $\sim 10^{-15}$  s, the observed instantaneous atomic correlation functions with anisotropy can be safely attributed to local orthorhombicity and nematic fluctuations at the length scale of  $\sim 1$  nm, that are observable also at much higher temperature and in an optimally doped  $\text{Ba}_{0.6}\text{K}_{0.4}\text{Fe}_2\text{As}_2$  superconductor.

It should be recalled that the local orthorhombicity tends to increase at lower temperature (Fig. 3), however, without any apparent anomaly. Instead, the valence electronic structure anisotropy shows an abrupt increase below  $\sim 100$  K (Fig. 6) which is rather close to the  $T_S$  or  $T_N$  of  $\text{BaFe}_2\text{As}_2$ . The fact that electronic nematic phase is known to appear around  $\sim 100$  K, we do not rule out additional triggers to affect the fluctuating nematicity driven by the local orthorhombicity in  $\text{Ba}_{0.6}\text{K}_{0.4}\text{Fe}_2\text{As}_2$ . One of the possible triggers can be magnetoelastic coupling that is known to play an important role in the iron-based superconductors. The importance of magnetoelastic coupling has been shown by a variety of experiments in these materials.<sup>14,15</sup> Indeed, it is common to find indicators of nematicity in the magnetic response function. Here, we have determined the local magnetic moment as a function of temperature in  $\text{Ba}_{0.6}\text{K}_{0.4}\text{Fe}_2\text{As}_2$  to explore a possible correlation between the magnetic response and the fluctuating nematic phase. The local magnetic moment has been determined by Fe K $\beta$  XES measurements performed on the same sample.

Fig. 7(a) shows the Fe K $\beta_{1,3}$  XES ( $3p \rightarrow 1s$ ) of  $\text{Ba}_{0.6}\text{K}_{0.4}\text{Fe}_2\text{As}_2$  at 300 K compared with the spectra of three references:  $\text{Fe}_2\text{O}_3$  carrying magnetic moment of  $\sim 4.6\mu_B$ , the non-magnetic  $\text{FeS}_2$  and metallic iron carrying magnetic moment of  $\sim 0.9\mu_B$ . The spectra are shown normalized with respect to the total integrated area. The Fe K $\beta$  XES probes the local magnetic moment due to (3p, 3d) exchange interactions,<sup>41</sup> modifying the K $\beta$  XES profile and splitting between the main K $\beta_{1,3}$  emission line and

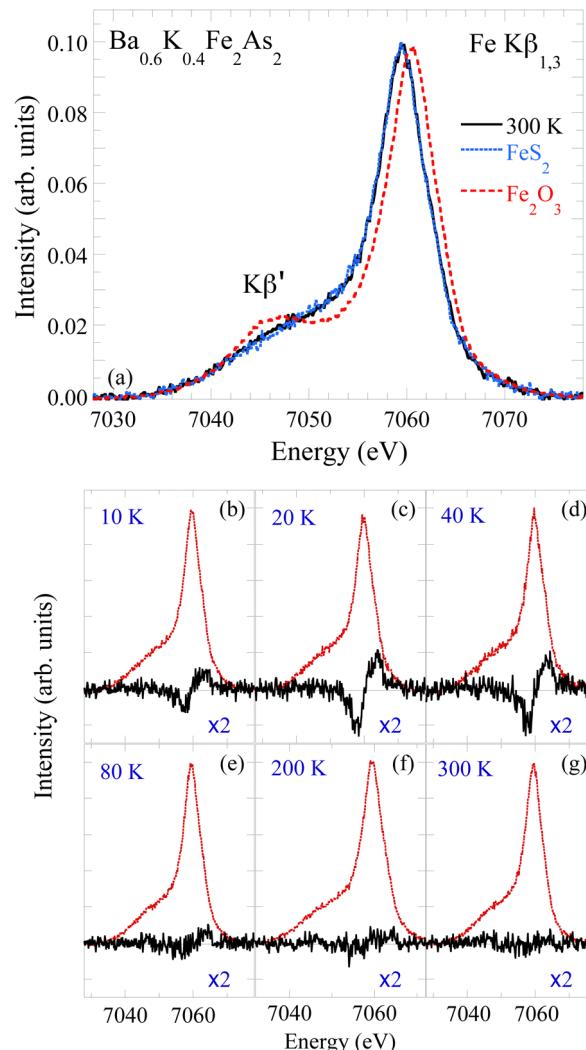
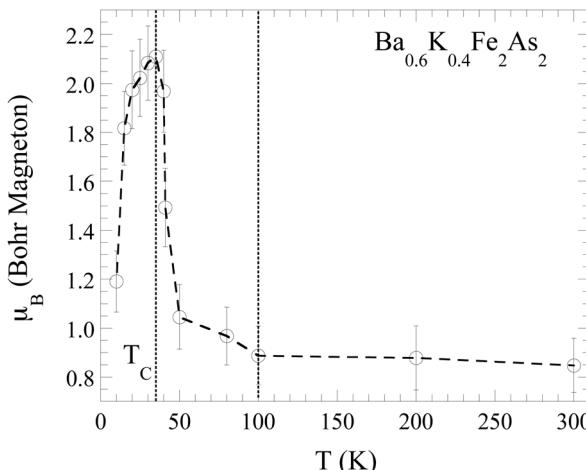


Fig. 7 (a) Integrated area normalized Fe K $\beta$  X-ray emission spectra of  $\text{Ba}_{0.6}\text{K}_{0.4}\text{Fe}_2\text{As}_2$  at 300 K plotted with the reference samples  $\text{FeS}_2$  and  $\text{Fe}_2\text{O}_3$ . (b)–(g) The Fe K $\beta$  XES (red) spectral differences (black) are shown (multiplied by two) at several temperatures.

the satellite K $\beta'$ . The K $\beta'$  spectral weight as well as the K $\beta_{1,3}$  energy increases with increasing 3d spin magnetic moment.<sup>41</sup> Thus the energy position of the K $\beta_{1,3}$  provides similar information on the magnetic moment as the K $\beta'$  satellite weight, reflecting the effective number of unpaired 3d electrons.<sup>41</sup> The XES spectra reveal non-negligible change as a function of temperature due to differing local Fe magnetic moment.<sup>41,60–64</sup> The local Fe magnetic moment  $\mu_B$  can be quantitatively evaluated from the integrated area of the absolute XES intensity difference (IAD) with respect to a nonmagnetic reference sample (ref. 62–64). Fig. 7(b)–(g) show the XES on  $\text{Ba}_{0.6}\text{K}_{0.4}\text{Fe}_2\text{As}_2$  at several temperatures together with the differences with respect to the one of  $\text{FeS}_2$ . The difference spectra show a clear evolution by lowering the sample temperature from 300 K.

Fig. 8 shows the temperature dependence of the local magnetic moment estimated by the IAD procedure and comparison with the known magnetic moments of the three references.



**Fig. 8** Integrated absolute difference (IAD) with respect to the spectrum of  $\text{FeS}_2$  at 300 K is shown as a function of temperature. The dashed line is a smooth fit to guide the eyes. The uncertainty is estimated by analyzing different XES scans at each temperature.

The magnetic moment at room temperature is  $\sim 0.9\mu_B$ , consistent with the magnetic moment reported earlier in the XES studies,<sup>63,64</sup> albeit slightly different from the fluctuating moment reported by neutron scattering.<sup>5</sup> By cooling down, the local moment shows an increase below  $\sim 100$  K, before showing a drop in the vicinity of the superconducting transition temperature. Considering the above, the increasing local magnetic moment below  $\sim 100$  K is likely to be associated with the nematic fluctuations in  $\text{Ba}_{0.6}\text{K}_{0.4}\text{Fe}_2\text{As}_2$ . On the other hand, decrease below the superconducting transition temperature is an indication of quenching of the local Fe magnetic moment, similar to the one found in other iron-based superconductors.<sup>65–67</sup> The suppression of the magnetic moment is also suggestive of its role in the superconductivity of  $\text{Ba}_{0.6}\text{K}_{0.4}\text{Fe}_2\text{As}_2$ .

Looking at the electronic anisotropy (Fig. 6), it seems that the local magnetic moment correlates with the nematic fluctuations in  $\text{Ba}_{0.6}\text{K}_{0.4}\text{Fe}_2\text{As}_2$ . It is interesting to note that, while the local orthorhombicity hardly shows any anomalous change, the linear dichroism of the 1s to 3d electronic transition (in the XANES) and the local magnetic moment (in the XES), both showing a clear increase  $\sim 100$  K. Therefore, the system may host two kinds of nematicity, *i.e.*, the former is due to orbital-lattice coupling while the latter is driven by the spin-lattice coupling. Nevertheless, the results suggest strong coupling of lattice, charge and magnetic degrees of freedom to drive nematic fluctuations in the iron-based superconductors.

## 4 Summary and conclusions

In summary, we have studied the anisotropic local structure of an optimally doped  $\text{Ba}_{0.6}\text{K}_{0.4}\text{Fe}_2\text{As}_2$  superconducting single crystal by polarized XAFS measurements performed with polarization of the X-ray beam parallel to the two in-plane directions ( $E \parallel a$  and  $E \parallel b$ ). The local structure is found to be anisotropic with differing Fe-As and Fe-Fe bond lengths and their stiffness.

Unlike the known average structure, the local structure shows clear traces of lower symmetry  $C_2$  phase in  $\text{Ba}_{0.6}\text{K}_{0.4}\text{Fe}_2\text{As}_2$  even at 300 K suggesting local and instantaneous orthorhombicity at the length scale of  $\sim 1$  nm. It is interesting to note that, using the experimental approach based on the polarized XAFS spectroscopy we are able to uncover the intrinsic anisotropy without using any external stimulant such as uniaxial pressure. The local orthorhombicity is consistent with the XANES results providing direct information on the local geometry and the valence electronic states. The electronic anisotropy due to differing orbital hybridization shows an anomalous change below  $\sim 100$  K where the commonly discussed nematic fluctuations occur. Furthermore, the local iron-magnetic moment measured by Fe K $\beta$  XES shows an increase around the same temperature followed by a decrease in the proximity of the superconducting transition temperature. The results suggest that the observed nematic fluctuations are likely to be coupled with the magnetic fluctuations in the optimally doped iron-based superconductors. In conclusion, the combination of single crystal sample and polarization dependence of the X-ray absorption based spectroscopy is an efficient tool to probe instantaneous local orthorhombicity and related exotic phenomena in quantum materials.

## Author contributions

N. L. S., H. E. and T. M. Conceptualization. L. T., G. D., S. I., A. I., L. S., C. M., H. E. and N. L. S. Investigations. L. T., G. T., G. M. P., M. Y. H. and N. L. S. Data curation and formal analysis. L. T. and N. L. S. Writing, review and editing. All authors discussed the results and commented on the manuscript.

## Data availability

The data that support the findings of this study are available from the corresponding author, [N. L. S.], upon reasonable request.

## Conflicts of interest

There are no conflicts to declare.

## Acknowledgements

We thank the ALBA and Elettra staffs for the assistance during the measurements. We also thank G. Aquilanti for discussions and providing useful feedback on the manuscript. The work is partially supported by the Sapienza University of Rome and JSPS KAKENHI (No. JP19H05823).

## References

- 1 A. Martinelli, F. Bernardini and S. Massidda, *C. R. Phys.*, 2016, **17**, 5–35.



2 Y. Gu, Z. Liu, T. Xie, W. Zhang, D. Gong, D. Hu, X. Ma, C. Li, L. Zhao, L. Lin, Z. Xu, G. Tan, G. Chen, Z. Y. Meng, Y.-F. Yang, H. Luo and S. Li, *Phys. Rev. Lett.*, 2017, **119**, 157001.

3 X. Luo and X. Chen, *Sci. China Mater.*, 2015, **58**, 77.

4 J. Paglione and R. Greene, *Nat. Phys.*, 2010, **6**, 645–658.

5 P. Dai, *Rev. Mod. Phys.*, 2015, **87**, 855.

6 D. J. Singh, *Phys. C*, 2009, **469**, 418–424.

7 K. Kuroki, S. Onari, R. Arita, H. Usui, Y. Tanaka, H. Kontani and H. Aoki, *New J. Phys.*, 2009, **11**, 025017.

8 A. Patz, T. Li and S. Ran, *et al.*, *Nat. Commun.*, 2014, **5**, 3229.

9 A. A. Kordyuk, *Low Temp. Phys.*, 2012, **38**, 888.

10 W. J. Meese, T. Vojta and R. M. Fernandes, *Phys. Rev. B*, 2022, **106**, 115134.

11 R. M. Fernandes, A. V. Chubukov and J. Schmalian, *Nat. Phys.*, 2014, **10**, 97.

12 Q.-P. Ding, W. R. Meier, J. Cui, M. Xu, A. E. Böhmer, S. L. Bud'ko, P. C. Canfield and Y. Furukawa, *Phys. Rev. Lett.*, 2018, **121**, 137204.

13 R. Stadel, D. D. Khalyavin, P. Manuel, K. Yokoyama, S. Lapidus, M. H. Christensen, R. M. Fernandes, D. Phelan, D. Y. Chung, R. Osborn, S. Rosenkranz and O. Chmaissem, *Commun. Phys.*, 2022, **5**, 146.

14 A. E. Böhmer, J.-H. Chu, S. Lederer and M. Yi, *Nat. Phys.*, 2022, **18**, 1412.

15 J. Wang, Y. Wu, X. Zhou, Y. Li, B. Teng, P. Dong, J. He and Y. Li, *Adv. Phys.: X*, 2021, **6**, 1878931, DOI: [10.1080/23746149.2021.1878931](https://doi.org/10.1080/23746149.2021.1878931).

16 Y. Gallais and I. Paul, *C. R. Phys.*, 2016, **17**, 113–139.

17 J. Li, P. J. Pereira, J. Yuan, Y.-Y. Lv, M.-P. Jiang, D. Lu, Z.-Q. Lin, Y.-J. Liu, J.-F. Wang, L. Li, X. Ke, G. V. Tendeloo, M.-Y. Li, H.-L. Feng, T. Hatano, H.-B. Wang, P.-H. Wu, K. Yamaura, E. Takayama-Muromachi, J. Vanacken, L. F. Chibotaru and V. V. Moshehalkov, *Nat. Commun.*, 2017, **8**, 1880.

18 X. Lu, J. T. Park, R. Zhang, H. Luo, A. H. Nevidomskyy, Q. Si and P. Dai, *Science*, 2014, **345**, 657.

19 A. E. Bohmer, F. Hardy, L. Wang, T. Wolf, P. Schweiss and C. Meingast, *Nat. Commun.*, 2015, **6**, 7911.

20 J. L. Niedziela, M. A. McGuire and T. Egami, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2012, **86**, 174113.

21 S. Li, M. Toyoda, Y. Kobayashi, M. Itoh, K. Ikeuchi, Y. Yoneda, A. Otani, D. Matsumura, S. Asano, J. Mizuki and M. Sato, *Phys. C*, 2018, **555**, 45.

22 X. Chen, S. Maiti, R. M. Fernandes and P. J. Hirshfeld, *Phys. Rev. B*, 2020, **102**, 184512.

23 S. Wu, Y. Song, Y. He, A. Frano, M. Yi, X. Chen, H. Uchiyama, A. Alatas, A. H. Said, L. Wang, T. Wolf, C. Meingast and R. J. Birgeneau, *Phys. Rev. Lett.*, 2021, **126**, 107001.

24 C. Wuttke, F. Cagliaris, S. Sykora, F. Steckel, X. Hong, S. Ran, S. Khim, R. Kappenberger, S. L. Bud'ko, P. C. Canfield, S. Wurmehl, S. Aswartham, B. Buechner and C. Hess, *npj Quantum Mater.*, 2022, **7**, 82.

25 K. Ishida, M. Tsujii, S. Hosoi, Y. Mizukami, S. Ishida, A. Iyo, H. Eisaki, T. Wolf, K. Grube, H. v Lohneysenc, R. M. Fernandes and T. Shibauchi, *Proc. Natl. Acad. Sci. U. S. A.*, 2020, **117**, 6424–6429.

26 L. Liu, T. Mikami, S. Ishida, K. Koshiishi, K. Okazaki, T. Yoshida, H. Suzuki, M. Horio, L. C. C. Ambolode, II, J. Xu, H. Kumigashira, K. Ono, M. Nakajima, K. Kihou, C. H. Lee, A. Iyo, H. Eisaki, T. Kakeshita, S. Uchida and A. Fujimori, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2015, **92**, 094503.

27 M. Nakajima, S. Ishida, Y. Tomioka, K. Kihou, C. H. Lee, A. Iyo, T. Ito, T. Kakeshita, H. Eisaki and S. Uchida, *Phys. Rev. Lett.*, 2012, **109**, 217003.

28 J. Hecher, S. Ishida, D. Song, H. Ogino, A. Iyo, H. Eisaki, M. Nakajima, D. Kagerbauer and M. Eisterer, *Phys. Rev. B*, 2018, **97**, 014511.

29 S.-F. Wu, W.-L. Zhang, V. K. Thorsmolle, G. F. Chen, G. T. Tan, P. C. Dai, Y. G. Shi, C. Q. Jin, T. Shibauchi, S. Kasahara, Y. Matsuda, A. S. Sefat, H. Ding, P. Richard and G. Blumberg, *Phys. Rev. Res.*, 2020, **2**, 033140.

30 I. R. Fisher, L. Degiorgi and Z. X. Shen, *Rep. Prog. Phys.*, 2011, **74**, 124506.

31 L. Chen, T. T. Han, C. Cai, Z. G. Wang, Y. D. Wang, Z. M. Xin and Y. Zhang, *Phys. Rev. B*, 2021, **104**, L060502.

32 K. Momma and F. Izumi, *J. Appl. Crystallogr.*, 2011, **44**, 1272.

33 B. A. Frandsen, K. M. Taddei, M. Yi, A. Frano, Z. Guguchia, R. Yu, Q. Si, D. E. Bugaris, R. Stadel, R. Osborn, S. Rosenkranz, O. Chmaissem and R. J. Birgeneau, *Phys. Rev. Lett.*, 2017, **119**, 187001.

34 B. A. Frandsen, K. M. Taddei, D. E. Bugaris, R. Stadel, M. Yi, A. Acharya, R. Osborn, S. Rosenkranz, O. Chmaissem and R. J. Birgeneau, *Phys. Rev. B*, 2018, **98**, 180505.

35 W. Wang, Y. Song, C. Cao, K.-F. Tseng, T. Keller, Y. Li, L. W. Harriger, W. Tian, S. Chi, R. Yu, A. H. Nevidomskyy and P. Dai, *Nat. Commun.*, 2018, **9**, 3128.

36 B. Karki, A. Alfailakawi, B. A. Frandsen, M. S. Everett, J. C. Neuefeind, B. Xu, H. Wang, M. Fang and B. Freelon, *Phys. Rev. B*, 2021, **104**, 064101.

37 B. Freelon, R. Sarkar, S. Kamusella, F. Brueckner, V. Grinenko, S. Acharya, M. Laad, L. Craco, Z. Yamani, R. Flacau, I. Swainson, B. Frandsen, R. Birgeneau, Y. Liu, B. Karki, A. Alfailakawi, J. C. Neuefeind, M. Everett, H. Wang, B. Xu, M. Fang and H.-H. Klauss, *npj Quantum Mater.*, 2021, **6**, 4.

38 D. C. Koningsberger and R. Prins, X-ray Absorption: Principles, Applications, Techniques of EXAFS, SEXAFS and XANES, *Chemical Analysis: A Series of Monographs on Analytical Chemistry and Its Applications*, Wiley, 1988.

39 G. Bunker, *Introduction to XAFS*, Cambridge University Press, 2010.

40 See, e.g. a review by J. J. Rehr and R. C. Albers, *Rev. Mod. Phys.*, 2000, **72**, 621.

41 P. Glatzel and U. Bergmann, *Coord. Chem. Rev.*, 2005, **249**, 65.

42 K. Kihou, T. Saito, K. Fujita, S. Ishida, M. Nakajima, K. Horigane, H. Fukazawa, Y. Kohori, S. Uchida, J. Akimitsu, A. Iyo, C.-H. Lee and H. Eisaki, *J. Phys. Soc. Jpn.*, 2016, **85**, 034718.

43 A. Di Cicco, G. Aquilanti, M. Minicucci, E. Principi, N. Novello, A. Cognigni and L. Olivi, *J. Phys.: Conf. Ser.*, 2009, **190**, 012043.



44 L. Simonelli, C. Marini, W. Olszewski, M. Avila Perez, N. Ramanan, G. Guilera, V. Cuartero and K. Klementiev, *Cogent Phys.*, 2016, **3**, 1231987.

45 L. Simonelli, C. Marini, L. Ribo, R. Homs, J. Avila, D. Heinis, I. Preda and K. Klementiev, *J. Synchrotron Radiat.*, 2023, **30**, 235–241.

46 B. Ravel and M. Newville, *Phys. Scr.*, 2006, **115**, 1007–1010.

47 S. J. Gurman, *J. Synchrotron Radiat.*, 1995, **2**, 56.

48 A. L. Ankudinov, B. Ravel, J. J. Rehr and S. D. Conradson, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 1998, **58**, 7565.

49 T. Ressler, *J. Synchrotron Radiat.*, 1998, **5**, 118.

50 D. Rybicki, M. Sikora, J. Stępień, L. Gondek, K. Goc, T. Strączek, M. Jurczyszyn, C. Kapusta, Z. Bukowski, M. Babij, M. Matusiak and M. Zajac, *Phys. Rev. B*, 2020, **102**, 195126.

51 B. Joseph, A. Iadecola, L. Simonelli, Y. Mizuguchi, Y. Takano, T. Mizokawa and N. L. Saini, *J. Phys.: Condens. Matter*, 2010, **22**, 485702.

52 S. Avci, O. Chmaissem, D. Y. Chung, S. Rosenkranz, E. A. Goremychkin, J.-P. Castellan, I. S. Todorov, J. A. Schlueter, H. Claus, A. Daoud-Aladine, D. D. Khalyavin, M. G. Kanatzidis and R. Osborn, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2012, **85**, 184507.

53 N. L. Saini, A. Lanzara, A. Bianconi and H. Oyanagi, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 1998, **58**, 11768.

54 N. L. Saini, H. Oyanagi, M. Molle, K. B. Garg, C. Kim and A. Bianconi, *J. Phys. Chem. Solids*, 2004, **65**, 1439.

55 E. Sevillano, H. Meuth and J. Rehr, *Phys. Rev. B: Solid State*, 1979, **20**, 4908.

56 A. G. de Figueiredo, M. R. Cantarino, W. R. da Silva Neto, K. R. Pakuszewski, R. Grossi, D. S. Christovam, J. C. Souza, M. M. Piva, G. S. Freitas, P. G. Pagliuso, C. Adriano and F. A. Garcia, *Phys. Rev. B*, 2022, **105**, 045130.

57 M. Hirano, Y. Yamada, T. Saito, R. Nagashima, T. Konishi, T. Toriyama, Y. Ohta, H. Fukazawa, Y. Kohori, Y. Furukawa, K. Kihou, C.-H. Lee, A. Iyo and H. Eisaki, *J. Phys. Soc. Jpn.*, 2012, **81**, 054704.

58 S.-F. Wu, P. Richard, H. Ding, H.-H. Wen, G. Tan, M. Wang, C. Zhang, P. Dai and G. Blumberg, *Phys. Rev. B*, 2017, **95**, 085125.

59 P. Wiecki, M. Frachet, A.-A. Haghaghirad, T. Wolf, C. Meingast, R. Heid and A. E. Bohmer, *Nat. Commun.*, 2021, **12**, 4824.

60 R. D. Cowan, *The Theory of Atomic Structure and Spectra*, University of California Press, Berkeley, (1981).

61 G. Vankó, T. Neisius, G. Molnár, F. Renz, S. Kárpáti, A. Shukla and F. M. F. de Groot, *J. Phys. Chem. B*, 2006, **110**, 11647.

62 S. Lafuerza, A. Carlantuono, M. Retegan and P. Glatzel, *Inorg. Chem.*, 2020, **59**, 12518.

63 H. Gretarsson, S. R. Saha, T. Drye, J. Paglione, J. Kim, D. Casa, T. Gog, W. Wu, S. R. Julian and Y.-J. Kim, *Phys. Rev. Lett.*, 2013, **110**, 047003.

64 H. Gretarsson, A. Lupascu, J. Kim, D. Casa, T. Gog, W. Wu, S. R. Julian, Z. J. Xu, J. S. Wen, G. D. Gu, R. H. Yuan, Z. G. Chen, N.-L. Wang, S. Khim, K. H. Kim, M. Ishikado, I. Jarrige, S. Shamoto, J.-H. Chu, I. R. Fisher and Y.-J. Kim, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2011, **84**, 100509.

65 L. Simonelli, T. Mizokawa, M. Moretti Sala, H. Takeya, Y. Mizuguchi, Y. Takano, G. Garbarino, G. Monaco and N. L. Saini, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2014, **90**, 214516.

66 F. Stramaglia, G. M. Pugliese, L. Tortora, L. Simonelli, C. Marini, W. Olszewski, S. Ishida, A. Iyo, H. Eisaki, T. Mizokawa and N. L. Saini, *J. Phys. Chem. C*, 2021, **125**, 10810.

67 G. Tomassucci, L. Tortora, G. M. Pugliese, F. Stramaglia, L. Simonelli, C. Marini, K. Terashima, T. Wakita, S. Ayukawa, T. Yokoya, K. Kudo, M. Nohara, T. Mizokawa and N. L. Saini, *Phys. Chem. Chem. Phys.*, 2023, **25**, 6684–6692.

