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# Constructing hollow core–shell Z-scheme heterojunction CdS@CoTiO<sub>3</sub> nanorods for enhancing the photocatalytic degradation of 2,4-DCP and TC

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Constructing Z-scheme heterojunctions incorporating an exquisite hollow structure is an effective performance regulation strategy for the realization of high quantum efficiency and a strong redox ability over photocatalysts. Herein, we report the delicate design and preparation of a core–shell hollow CdS@CoTiO<sub>3</sub> Z-scheme heterojunction with a CdS nanoparticle (NP)-constructed outer shell supported on a CoTiO<sub>3</sub> nanorod (NR) inner shell. The *in situ* growth synthetic method led to a tightly connected interface for the heterojunction between CdS and CoTiO<sub>3</sub>, which shortened the transport distance of photoinduced charges from the interface to the surface. The promoted charge carrier separation efficiency and the retained strong redox capacity caused by the Z-scheme photoinduced charge-transfer mechanism were mainly responsible for the boosted photocatalytic performance. Additionally, the well-designed core–shell structure afforded a larger interfacial area by the multiple direction contact between CdS and CoTiO<sub>3</sub>, ensuring sufficient channels for efficient charge transfer, and thus further boosting the photocatalytic activity. As an efficient photocatalyst, the optimized CdS@CoTiO<sub>3</sub> nano hybrids displayed excellent 2,4-dichlorophenol (2,4-DCP) and tetracycline (TC) degradation efficiencies of 91.3% and 91.8%, respectively. This study presents a Z-scheme heterojunction based on ecofriendly CoTiO<sub>3</sub>, which could be valuable for the development of metal perovskite photocatalysts for application in environmental remediation, and also demonstrated the tremendous potential of integrating a Z-scheme heterojunction with the morphology design of photocatalysts.

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## 1. Introduction

The photocatalytic remediation of waste effluent containing toxic organic pollutants is a promising strategy to mitigate the ever-increasing environmental pollution crisis owing to its satisfactory catalytic efficiency and ecofriendly features. However, the practical application of photocatalysis is still limited due to most current photocatalysts suffering from poor visible-light-absorption performance and a fast recombination of photoinduced carriers.<sup>1–3</sup> Developing efficient photocatalysts is highly desirable to promote the practical implementation of photocatalysis. Rationally constructing a heterojunction is an

effective strategy to accelerate the spatial separation of photoinduced carriers in semiconductors and the supply of more potential catalytic sites compared to in the single-component counterparts.<sup>4,5</sup> In particular, by forming a Z-scheme heterojunction, a wide light-absorption range and sufficient redox capacity can be simultaneously achieved, ascribed to the Z-scheme charge-transfer path with the assistance of the built-in electric field in the heterojunction region, which can thus enhance the photocatalytic performance.<sup>6–9</sup> Recently, a number of Z-scheme heterojunctions such as g-C<sub>3</sub>N<sub>4</sub>/MoS<sub>2</sub>,<sup>10</sup> CdIn<sub>2</sub>S<sub>4</sub>/ZnS,<sup>11</sup> CoSx/CdS,<sup>12</sup> and BiOCl/Bi<sub>2</sub>S<sub>3</sub><sup>13</sup> have been constructed with the aim of achieving excellent photocatalytic performances. Additionally, assembling low-dimensional nanoscale building blocks to form hierarchical hollow configurations is reported as another commendable method for improving the photocatalytic performance.<sup>14,15</sup> This is mainly because hierarchical hollow structures possess multiple intrinsic advantages, including improved light absorption, expedited photoinduced carrier separation and migration, and enhanced surface redox reactions.<sup>16–20</sup> So far, numerous nanomaterials

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with hierarchical hollow configurations have been developed as photocatalysts to catalyze  $\text{CO}_2$  conversion,<sup>21</sup> water splitting reactions,<sup>22</sup> and the photocatalytic degradation of pollutants.<sup>23</sup> Z-scheme heterojunction photocatalysts can further accelerate the separation and direct transfer of light-induced electron-hole pairs to promote the redox reactions. It is noteworthy that forming core-shell structures with distinct boundaries can not only provide larger active interfaces between the constituent materials by multiple directional contact, but also produce new synergistic effects owing to the ranges of physical and chemical properties of the core and shell influenced by their compositions, structures, and dimensions, making them attractive for applications in catalysis.<sup>24,25</sup> For example, Zhu *et al.*<sup>26</sup> prepared a  $\text{Sn}^{2+}$ -ion self-doping  $\text{CdS}/\text{ZnSnO}_3$  hollow core-shell cubic heterojunction photocatalyst, which demonstrated significantly improved photocatalytic activity for the hydrogen evolution reaction and for ciprofloxacin degradation under visible-light irradiation.

Among the numerous reported photocatalysts, perovskite oxide  $\text{CoTiO}_3$  has aroused much interest recently owing to its superior photoelectronic properties, visible-light-responsive band gap ( $E_g \approx 2.3$ ), and high carrier mobility. In particular, its unique  $\text{ABO}_3$  structure can increase ionic defects or oxygen vacancies *via* holding more cations, resulting in improved photocatalytic performances.<sup>27,28</sup> Importantly,  $\text{CoTiO}_3$  possess a high valence band potential ( $E_v \approx 2.3$ ) that endow  $E_v$  holes with strong oxidizability.  $\text{CoTiO}_3$ -based photocatalytic materials have already been exploited for pollutant degradation and have achieved excellent performances.<sup>29–31</sup> Apparently, the introduction of a complementary semiconductor with a strong reduction capacity to construct Z-scheme heterojunctions with  $\text{CoTiO}_3$  can not only contribute to promoting a strong redox ability but also fast carrier separation and transportation, thus further improving the photocatalytic performance.  $\text{CdS}$  with a moderate band gap ( $\approx 2.4$  eV) has been widely reported to be a remarkable photocatalyst.<sup>32–34</sup> Moreover,  $\text{CdS}$  possessing the required conduction band potential ( $E_c$ ) of about  $-0.5$  V is more negative than that of  $\text{CoTiO}_3$ , which makes it one of the potential candidates to construct Z-scheme heterojunctions with  $\text{CoTiO}_3$ . The fabrication of heterostructured nanocomposites of  $\text{CoTiO}_3$  by combining with  $\text{CdS}$  can not only endow the hybrids with novel characteristics, but also improve their

efficiency due to synergistic effects. The given background supports the feasibility of synergistic photocatalysis by combining  $\text{CoTiO}_3$  and  $\text{CdS}$ . Yet, to the best of our knowledge, the use of hybrids of  $\text{CdS}$  and  $\text{CoTiO}_3$  as an efficient photocatalyst for the degradation of toxic organic pollutants in waste effluents has not been reported previously.

To realize the ingenious synergistic advantages of a Z-scheme heterojunction and a hollow structure, a hierarchical hollow  $\text{CdS}@/\text{CoTiO}_3$  Z-scheme heterojunction was obtained by the *in situ* growth of  $\text{CdS}$  NPs on the surface of  $\text{CoTiO}_3$  NRs, in which the  $\text{CoTiO}_3$  NRs served as an outstanding matrix that could also greatly prevent the  $\text{CdS}$  NPs from agglomerating. The unique structure promoted the separation and transfer of photogenerated charges, while also offering a large surface area for target pollutants adsorption, and the exposure of abundant active sites for surface catalysis. Accordingly, the optimized  $\text{CdS}@/\text{CoTiO}_3$  hybrids manifested superior activity for the visible-light degradation of 2,4-DCP and TC, achieving remarkable degradation efficiencies of  $\sim 91.3\%$  (2,4-DCP) and  $91.8\%$  (TC), respectively. Additionally,  $\text{CdS}@/\text{CoTiO}_3$  exhibited remarkable stability over four successive cycles of the degradation process. Based on the experiment results, the enhanced photocatalytic degradation mechanism of  $\text{CdS}@/\text{CoTiO}_3$  is proposed.

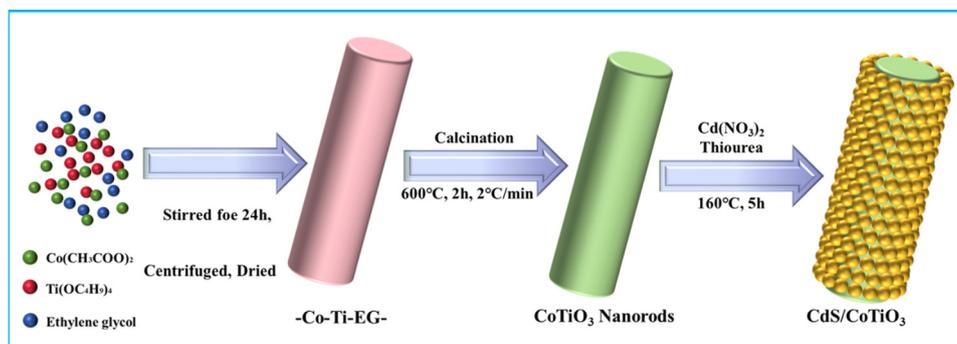
## 2. Experimental section

### 2.1. Materials

The key reagents were  $\text{Co}(\text{CH}_3\text{COOH})_2 \cdot 4\text{H}_2\text{O}$ , tetrabutyl titanate, ethylene glycol (EG), diethylenetriamine (DETA),  $\text{CdSO}_4 \cdot 8\text{H}_2\text{O}$ , thiourea, (Kemiou Chemical Reagent Co. Ltd, Tianjin, China), 2,4-dichlorophenol, and tetracycline (Aladdin Biological Co., Ltd, Shanghai, China). All the reagents were of analytical grade.

### 2.2. Synthesis

**2.2.1. Preparation of hollow  $\text{CoTiO}_3$  nanorods.** The  $\text{CoTiO}_3$  was prepared according to a literature method<sup>35,36</sup> with some modifications as follows:  $\text{Co}(\text{CH}_3\text{COOH})_2 \cdot 4\text{H}_2\text{O}$  (0.244 g) and butyl titanate (0.4 mL) were mixed with ethylene glycol (60 mL) and vigorously stirred at room temperature for 24 h to obtain light-green  $\text{Co-Ti-ethylene glycol}$  ( $\text{Co-Ti-EG-}$ ). The  $\text{Co-Ti-EG-}$  was collected by centrifugation. After washing with ethanol



Scheme 1 Illustration of the synthesis of the  $\text{CdS}@/\text{CoTiO}_3$  Z-scheme heterojunction.

three times, the obtained Co–Ti–EG– was dried in an oven at 80 °C and calcined at 600 °C for 2 h under vacuum conditions at a heating rate of 2 °C min<sup>-1</sup> to obtain the CoTiO<sub>3</sub> nanorod structures as a green powder.

### 2.2.2. Preparation of the core–shell CdS@CoTiO<sub>3</sub> hybrids.

The CdS@CoTiO<sub>3</sub> hybrid was synthesized by the *in situ* growth of CdS on the surface of CoTiO<sub>3</sub> *via* a facile solvothermal method, as illustrated in Scheme 1. Here, 10%CdS@CoTiO<sub>3</sub>, 0.0865 g Cd(NO<sub>3</sub>)<sub>2</sub>·4H<sub>2</sub>O, and 0.0215 g thiourea were added into an 80 mL mixed solvent of 64 mL deionized H<sub>2</sub>O and 16 mL DETA, and then a certain amount of CoTiO<sub>3</sub> (0.3724 g) was added as the substrate, with stirring for 5 min. After that, the above mixture was transferred to a 100 mL Teflon-lined stainless steel autoclave and kept at 160 °C for 5 h. Finally, the product was washed with water and ethanol three times, collected, and dried in an oven at 80 °C for 24 h. The obtained samples were denoted as *x*CdS@CoTiO<sub>3</sub> (where *x* means the mass ratio (%) of CdS relative to CoTiO<sub>3</sub>, and *x* = 5, 10, 15, and 20). For comparison, pure CdS was prepared by the same method in the absence of CoTiO<sub>3</sub>.

**2.2.3. Characterizations.** The morphologies and microstructures of the catalysts were analyzed by SEM (S4800, Hitachi Co., Japan), HRTEM coupled with EDX elemental scanning spectroscopy (S4800, Hitachi Co., Japan and 2100F, JEOL Co. Japan), XRD with Cu K $\alpha$  radiation (D/max-2200, Rigaku., Japan), and XPS (Thermo Electron, ESCALAB 250 Xi., USA). The UV-visible diffuse reflectance spectra and photoluminescence spectra were acquired using UV2550 and FLS-920T fluorescence spectrophotometers, respectively. The ESR signals were recorded using a Bruker A300 spectrometer. The photoelectrochemical and electrochemical impedance spectra were obtained on a CHI660D electrochemical workstation (Shanghai Chenhua, China).

### 2.2.4. Evaluation of the photocatalytic degradation activity.

Here, 2,4-DCP (10 mg L<sup>-1</sup>) and TC (10 mg L<sup>-1</sup>) were selected to estimate the photocatalytic activities of the obtained materials under visible-light irradiation. A 500 W Xenon lamp with a 420 nm cutoff filter (100 mW cm<sup>-2</sup> light intensity) was used as the visible-light source and the mixture was maintained at around 20 cm distance from it. First, 30 mg photocatalyst was added into a photocatalytic reactor containing 50 mL 2,4-DCP/TC solution and then placed in darkness under magnetic stirring for 40 min at 300 rpm to achieve adsorption–desorption equilibrium. After that, the light source was turned on and 2 mL aqueous was withdrawn from the reactor at certain time intervals, and centrifuged to remove the catalysts. The concentration of 2,4-DCP was monitored using an Agilent 1290 Infinity II ultraperformance liquid chromatography (UPLC) system equipped with a C18 column and UV detector ( $\lambda$  = 284 nm) and the mobile phase was HCOOH and CH<sub>3</sub>OH (50/50, *v:v*) with a flow rate of 1.0 mL min<sup>-1</sup>, while TC was analyzed using a Lambda 750S UV/VIS/NIR spectrometer at  $\lambda$  = 358 nm.

The degradation ratio (*R*) of 2,4-DCP and TC was calculated according to the following equation:

$$R\% = \frac{C_0 - C_t}{C_0} \times 100\% \quad (1)$$

where *C*<sub>0</sub> and *C*<sub>*t*</sub> denote the 2,4-DCP and TC concentrations in mg L<sup>-1</sup> at the initial time and at time *t* (min), respectively.

## 3. Results and discussion

It could be clearly observed that pristine CoTiO<sub>3</sub> displayed a rod-like morphology with a length and width of approximately 2  $\mu$ m and 400 nm, respectively (Fig. 1a), and a definite interior cavity could also be observed (see the inset of Fig. 1a). Meanwhile, the morphology analysis of pure CdS showed it was composed of a great many irregular nanoparticles agglomerated together (Fig. 1b). Following the hydrothermal reaction at 160 °C, CdS NPs uniformly grew on the whole surface of the CoTiO<sub>3</sub> NRs to form the shell layer without any agglomeration (Fig. 1c), meaning there was a large contact area between CdS and CoTiO<sub>3</sub> to form more heterojunctions. A typical TEM image of the as-derived sample further confirmed the hollow core-shell structure with a thickness of CdS shell layer of about 40 nm, revealing that the incorporation of CdS had no obvious influence on the morphology.

Additionally, HRTEM images for the CdS@CoTiO<sub>3</sub> heterojunction are shown in Fig. 2a and b. According to the partially magnified image, a clear heterojunction interface with the intersection of the lattice fringes could be observed, further confirming the forming of the CdS@CoTiO<sub>3</sub> heterojunction. The tight anchoring of CdS NPs on the CoTiO<sub>3</sub> nanorods favored the formation of intimate interfacial contacts, which is highly beneficial for efficient charge separation/transfer in hybrids.<sup>37</sup> The clear lattice fringe with a spacing of 0.37 nm were attributed to the (012) plane of CoTiO<sub>3</sub> nanorods (Fig. 2c), while that of 0.34 nm was assigned to the (002) plane of CdS (Fig. 2d).<sup>38,39</sup> To further obtain information on the elemental composition and distribution for the CdS@CoTiO<sub>3</sub> heterojunction photocatalyst, scanning TEM-energy-dispersive X-ray (STEM-EDX) characterization was conducted. As shown in Fig. 2e, it could be observed that the CdS@CoTiO<sub>3</sub> heterojunction contained Co, Ti, O, Cd, and S elements, and their distribution was homogeneous, supporting the results obtained from the TEM and HRTEM

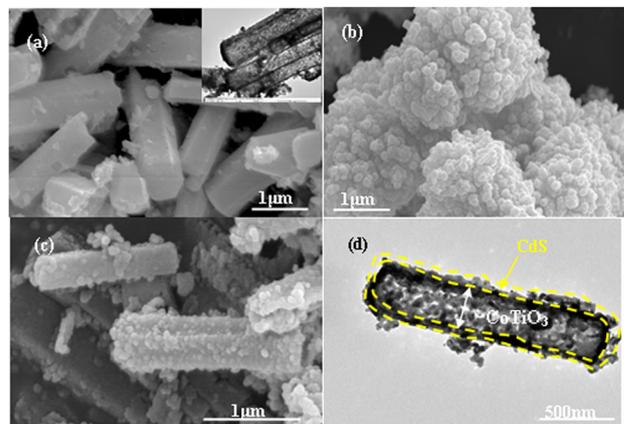


Fig. 1 SEM images: CoTiO<sub>3</sub> (inset is the TEM image) (a), CdS (b), CdS@CoTiO<sub>3</sub> (c); and TEM image of CdS@CoTiO<sub>3</sub> (d).

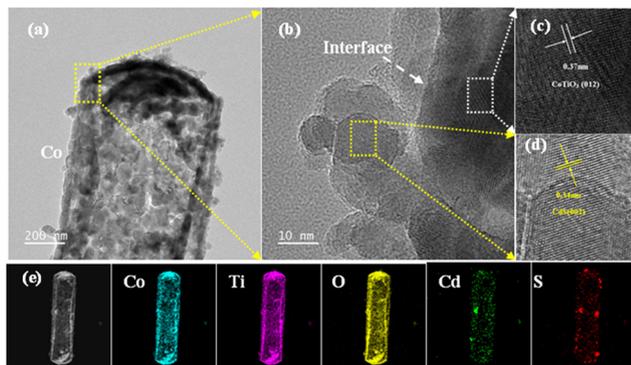


Fig. 2 (a)–(d) HRTEM micrographs and (e) corresponding EDS elemental mapping images of the CdS@CoTiO<sub>3</sub> heterojunction.

studies showing that the CdS nanoparticles were uniformly covered on the CoTiO<sub>3</sub> core nanostructures.

XPS was performed to obtain information on the elements present and on the alteration in the electronic environments of the different samples. The XPS spectra of CoTiO<sub>3</sub>, CdS, and CdS@CoTiO<sub>3</sub> verified the existence of the corresponding elements (Fig. 3a and b). Fig. 3c illustrates the fine spectra of Co in CoTiO<sub>3</sub> and CdS@CoTiO<sub>3</sub>. The two peaks at about 796.18 and 780.18 eV with their distinguishable satellite peaks (at about 802.38 and 785.68 eV, denoted as Sta.) were indexed to Co 2p<sub>3/2</sub> and Co 2p<sub>1/2</sub>. Moreover, the splitting energy between the Co 2p<sub>1/2</sub> and Co 2p<sub>3/2</sub> peaks of CoTiO<sub>3</sub> and CdS@CoTiO<sub>3</sub> were consistent with the theoretical value of 16 eV for Co(II), revealing the presence of Co<sup>2+</sup> in CoTiO<sub>3</sub>.<sup>27,40</sup> Notably, the binding energies were positively shifted by 0.60 eV for the CdS@CoTiO<sub>3</sub> heterojunction. The Ti 2p spectra for CoTiO<sub>3</sub> and CdS@CoTiO<sub>3</sub> are portrayed in Fig. 3d, where the peaks at about 457.51 and 463.24 eV were from the 2p<sub>3/2</sub> and 2p<sub>1/2</sub> orbitals of Ti, respectively.<sup>41,42</sup> After incorporating CdS NPs, the binding energy was 0.57 eV higher than that of pure CoTiO<sub>3</sub>. The high-resolution O 1s XPS spectrum of the CdS@CoTiO<sub>3</sub> heterojunction was divided into three peaks, in which the peak at 532.98 eV originated from the –OH groups adsorbed on the surface (Fig. 3e),<sup>43</sup> while the peaks at 530.47 and 529.16 eV represent the oxygen present in the lattice of CoTiO<sub>3</sub>.<sup>44,45</sup> A positive shift by around 0.20 eV was also observed in the CdS@CoTiO<sub>3</sub> heterojunction. At the same time, the binding energies for Cd 3d and S 2p for the CdS@CoTiO<sub>3</sub> heterojunction were negatively shifted of 0.20 and 0.50 eV with respect to the pristine CdS NPs, respectively (Fig. 3f and g). It is well accepted that the binding energy is negatively correlated with surface electron density.<sup>46</sup> The positive shifts of the Co 2p, Ti 2p, and O 1s XPS peaks as well as the negative shifts of the Cd 3d and S 2p regions on CdS@CoTiO<sub>3</sub> suggested that the electrons transfer from CoTiO<sub>3</sub> to CdS, further manifesting that hybrids with efficient electron-transfer processes were formed in the CdS@CoTiO<sub>3</sub> heterojunction.<sup>47–49</sup>

Fig. 3h presents the X-ray diffraction (XRD) patterns for CdS, CoTiO<sub>3</sub>, and a series of CdS@CoTiO<sub>3</sub> heterojunctions. For CdS and CoTiO<sub>3</sub>, all the diffraction peaks matched the respective

standard cards (CdS, PDF#10-0454)<sup>50</sup> and (CoTiO<sub>3</sub>, PDF#15-0866).<sup>51</sup> As for the CdS@CoTiO<sub>3</sub> heterojunctions, not only were diffraction peaks corresponding to CoTiO<sub>3</sub> observed, but also peaks for CdS. Interestingly, the intensity of the peaks for CdS gradually increased with the dosage increase, which unambiguously confirmed the successful synthesis of the CdS@CoTiO<sub>3</sub> heterojunction.

The light-absorption property was investigated by the UV-vis-DRS technique for better exploring the photocatalytic mechanism of the CdS@CoTiO<sub>3</sub> hybrids. As shown in Fig. 4a, the absorption edge of the CdS NPs was about 550 nm, concurring with related reports.<sup>13</sup> The pure CoTiO<sub>3</sub> NRs displayed optical adsorption properties both in the UV and visible-light regions. Furthermore, absorption peaks centered at about 538 and 605 nm were observed, which could be indexed to metal-to-metal charge transfer (Co<sup>2+</sup> to Ti<sup>4+</sup>).<sup>52</sup> Notably, after assembling CdS NPs on the CoTiO<sub>3</sub> hollow NRs, a stronger absorption intensity was discovered compared to CoTiO<sub>3</sub>, implying that the CdS@CoTiO<sub>3</sub> hybrid enabled efficient light harvesting, which was mainly due to the well-designed core-shell structure, wherein the tightly connected interface accelerated the transmission rate of charge carriers, and thus promoted the photocatalytic performance. According to the transformed Kubelka–Munk function,<sup>53</sup> the corresponding bandgaps of CdS, CoTiO<sub>3</sub>, and CdS@CoTiO<sub>3</sub> were estimated to be 2.23, 2.48, and 2.42, respectively (Fig. 4b).

Subsequently, the band structures of CoTiO<sub>3</sub> and CdS were calculated using the following equations:<sup>54,55</sup>

$$E_C = \chi - E_0 - 1/2E_g \quad (2)$$

$$E_V = E_g + E_C \quad (3)$$

where  $\chi$  and  $E_g$  are the bulk electronegativity and bandgap of the semiconductors, respectively, and  $E_0$  represents the energy of free electrons *versus* NHE ( $\approx 4.5$  eV). The estimated values of  $\chi$  for CoTiO<sub>3</sub> and CdS were 5.76 and 5.19 eV, respectively. Hence,  $E_C$  and  $E_V$  for CoTiO<sub>3</sub> were calculated to be 0.02 and 2.50 eV, respectively, while these of CdS were estimated to be –0.43 and 1.80 eV, respectively.

Various photoelectrochemical techniques were applied to obtain insights into the separation and transfer performance of the carriers between CdS and CoTiO<sub>3</sub> in the CdS@CoTiO<sub>3</sub> hybrid. Compared to CdS and CoTiO<sub>3</sub> alone, CdS@CoTiO<sub>3</sub> clearly displayed an enhancement in photoelectric current density (Fig. 5a), implying the lowest recombination rate of photogenerated carriers.<sup>56</sup> Moreover, the lower charge-transfer resistance in the CdS@CoTiO<sub>3</sub> hybrid was also confirmed by the smaller high-frequency semicircle in the electrochemical impedance spectroscopy results (EIS, Fig. 5b).<sup>57</sup> The photoelectrochemical characterizations indicated the accelerated separation and migration of photoinduced charges in the CdS@CoTiO<sub>3</sub> hybrid, and thus a prominent activity for photodegrading pollutants. Additionally, an obvious inhibition of the recombination of photogenerated e<sup>–</sup>/h<sup>+</sup> in the CdS@CoTiO<sub>3</sub> hybrids was indicated by the PL emission spectrum. Generally, a weaker PL intensity means a higher separation and transfer performance of photogenerated e<sup>–</sup>/h<sup>+</sup> pairs.<sup>17,58</sup> As presented

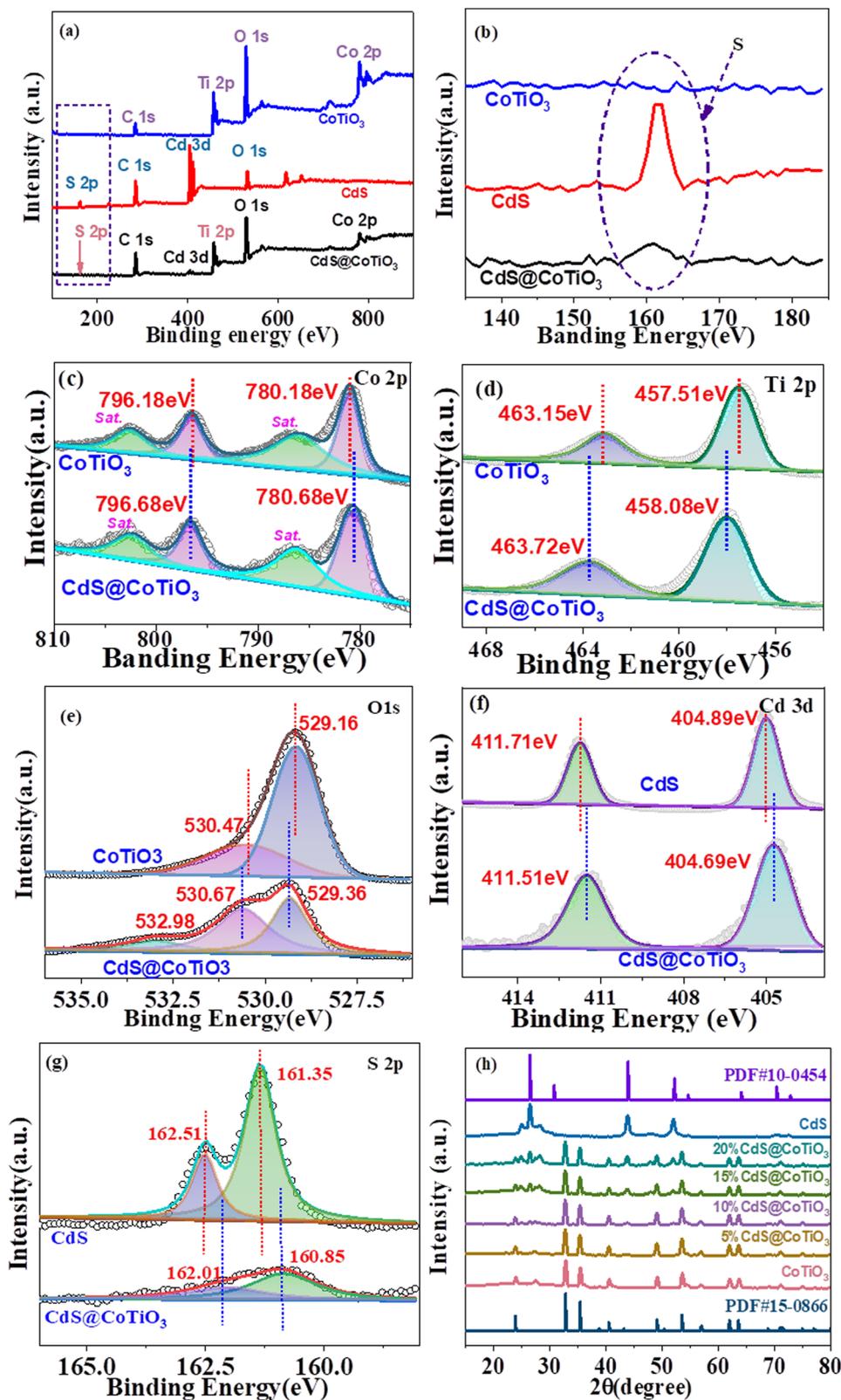


Fig. 3 (a) and (b) XPS survey spectra and high-resolution XPS spectra: (c) Co 2p, (d) Ti 2p, (e) O 1s, (f) Cd 3d, (g) S 2p; and (h) the XRD spectra of different samples.

in Fig. 5c, the emission intensity of the CdS@CoTiO<sub>3</sub> hybrids was remarkably weaker than those of CdS and CoTiO<sub>3</sub> alone,

signifying the recombination of light-induced charges in the CdS@CoTiO<sub>3</sub> hybrids was effectively prohibited.

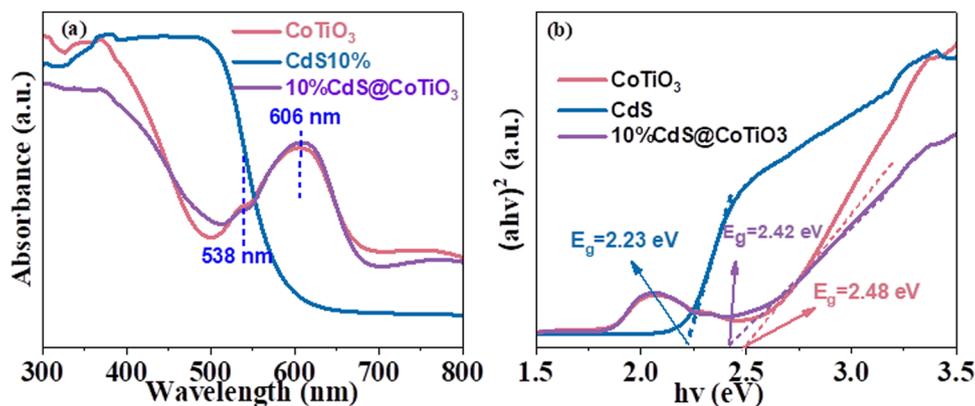


Fig. 4 (a) UV-vis-diffuse reflectance spectra and (b) corresponding  $(\alpha h\nu)^2$  versus  $h\nu$  curves of the pure  $\text{CoTiO}_3$ , CdS, and  $\text{CdS@CoTiO}_3$  heterojunction.

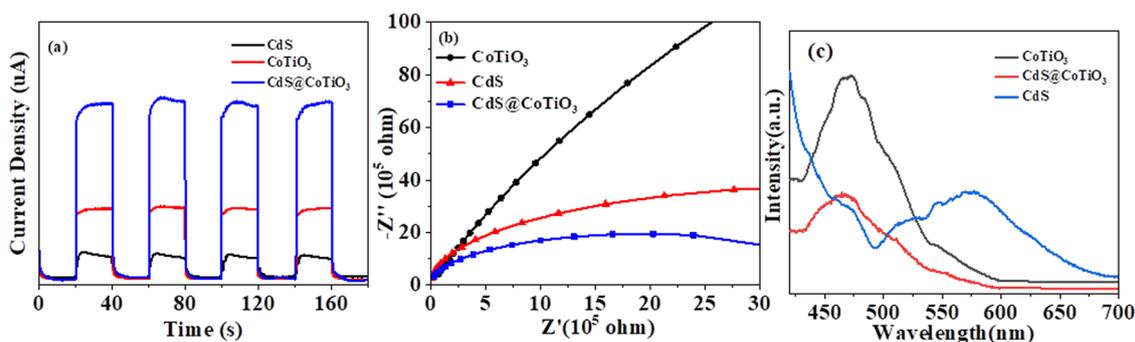


Fig. 5 (a) Visible-light photocurrent responses, (b) EIS Nyquist plots, and (c) PL plots of the pure  $\text{CoTiO}_3$ , CdS, and  $\text{CdS@CoTiO}_3$  heterojunction.

To evaluate the photocatalytic activity of the as-prepared catalysts, the representative organic pollutant 2,4-DCP solution was chosen as a pollution model, as exhibited in Fig. 6. Clearly, all the samples were essentially saturated by adsorption within 40 min in the dark, with only a small decrease in the concentration of 2,4-DCP. Moreover, only a negligible removal of 2,4-DCP was observed in the absence of any photocatalyst, suggesting that 2,4-DCP exhibited excellent stability under visible-light irradiation (Fig. 6a). All the  $\text{CdS@CoTiO}_3$  hybrids delivered boosted degradation capability toward 2,4-DCP compared with CdS and  $\text{CoTiO}_3$  alone, which was attributed to the acceleration

of the separation of photogenerated carriers and the strength of the redox capacity caused by the formation of a hollow core-shell Z-scheme heterojunction between CdS and  $\text{CoTiO}_3$ . Among the hybrids with different masses of CdS, 10%  $\text{CdS@CoTiO}_3$  showed the highest activity, with a removal efficiency of approximately 91.3%, while 50% and 46% removal efficiencies were obtained for CdS and  $\text{CoTiO}_3$  after 120 min visible irradiation. Fig. 6b displays the fitting of the photocatalytic degradation, whereby all the catalysts conformed to the quasi-first-order kinetic model. The kinetic constant ( $k$ ) of 10%  $\text{CdS@CoTiO}_3$  ( $0.018 \text{ min}^{-1}$ ) was about

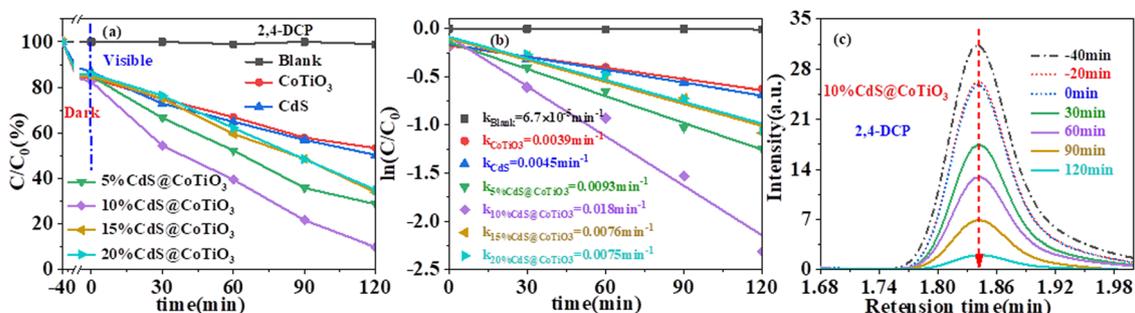


Fig. 6 (a) Photocatalytic degradation of 2,4-DCP aqueous solutions, and the (b) corresponding kinetic fitting curves over different samples under visible-light irradiation; and (c) corresponding UPLC plots of 2,4-DCP during degradation over the 10% $\text{CdS@CoTiO}_3$  hybrid.

4.0- and 4.6-times higher than those of pure CdS ( $0.0045 \text{ min}^{-1}$ ) and  $\text{CoTiO}_3$  ( $0.0039 \text{ min}^{-1}$ ), respectively. The concentration changes of 2,4-DCP was detected by UPLC. As depicted in Fig. 6c, the intensity of the peak for 2,4-DCP rapidly decreased with increasing the irradiation time, suggesting that 2,4-DCP was gradually decomposed over the 10%  $\text{CdS@CoTiO}_3$  heterojunction. Considering its most effective performance, 10%  $\text{CdS@CoTiO}_3$  was chosen for the subsequent research studies, without special instructions.

Subsequently, the photocatalytic degradation of a well-known antibiotic (TC) was investigated to confirm the enhancement of the photodegradation performance of the 10%  $\text{CdS@CoTiO}_3$  hybrids under identical conditions (Fig. 7). Obviously, the removal over the  $\text{CdS@CoTiO}_3$  hybrids ( $\approx 91.8\%$ ) was much higher than that for  $\text{CoTiO}_3$  alone ( $\approx 65.3\%$ ) with 80 min visible irradiation (Fig. 7a and b), with a  $k$  of  $0.025 \text{ min}^{-1}$ , which was 2.5-times higher than that of  $\text{CoTiO}_3$  ( $0.010 \text{ min}^{-1}$ ). Furthermore, the photocatalytic degradation of TC was confirmed by recording its UV-visible spectral behavior at  $\lambda_{\text{max}} = 358 \text{ nm}$  (Fig. 7c and d).

Generally, cycling stability is a crucial parameter to assess the industrialization potential of photocatalysts. Therefore, the 10%  $\text{CdS@CoTiO}_3$  hybrid with the best catalytic activity was selected for the cycling tests, as shown in Fig. 8. It could be clearly seen that the photocatalytic performance was not significantly attenuated for 2,4-DCP degradation after four cycling tests (Fig. 8a). Furthermore, the XRD spectrum of 10%

$\text{CdS@CoTiO}_3$  showed no obvious change after the reaction compared with that before, while the architectural structure of the  $\text{CdS@CoTiO}_3$  hybrids was also well maintained after the cycles (Fig. 8b and c), further implying the satisfactory reusability and stability of  $\text{CdS@CoTiO}_3$  during the photocatalytic reaction.

Next, trapping tests were performed to investigate the reactive species taking part in the photodegradation of 2,4-DCP process. Experimentally, isopropanol (IPA), sodium oxalate ( $\text{Na}_2\text{C}_2\text{O}_4$ ), and benzoquinone (BQ) were used to scavenge hydroxyl radicals ( $\cdot\text{OH}$ ), superoxide radicals ( $\cdot\text{O}_2^-$ ), and holes ( $\text{h}^+$ ), respectively.<sup>11</sup> As illustrated in Fig. 9a, with the addition of IPA and BQ, the degradation efficiency of 2,4-DCP significantly decreased from 91.0% to 49.5% and 45.1%, indicating that  $\cdot\text{OH}$  and  $\cdot\text{O}_2^-$  as primary active species participated in 2,4-DCP degradation. Meanwhile, the remove efficiency was also inhibited from 91.0% to 61.9% after introducing  $\text{Na}_2\text{C}_2\text{O}_4$ , implying that  $\text{h}^+$  also play a certain role in the photocatalytic degradation of 2,4-DCP. Furthermore, the generation of  $\cdot\text{OH}$  and  $\cdot\text{O}_2^-$  in the photocatalytic reaction system was further confirmed by EPR using DMPO as a free radical trapping agent, as exhibited in Fig. 9b. Clearly, no EPR signal was observed under the dark condition, indicating no  $\cdot\text{OH}$  or  $\cdot\text{O}_2^-$  were produced.<sup>59,60</sup> However, four characteristic signals of DMPO- $\cdot\text{OH}$  possessing the intensity ratio of 1:2:2:1 were observed in the EPR spectrum after 5 min of visible-light irradiation. Similarly, typical DMPO- $\cdot\text{O}_2^-$  adducts (quartet signal relative intensities

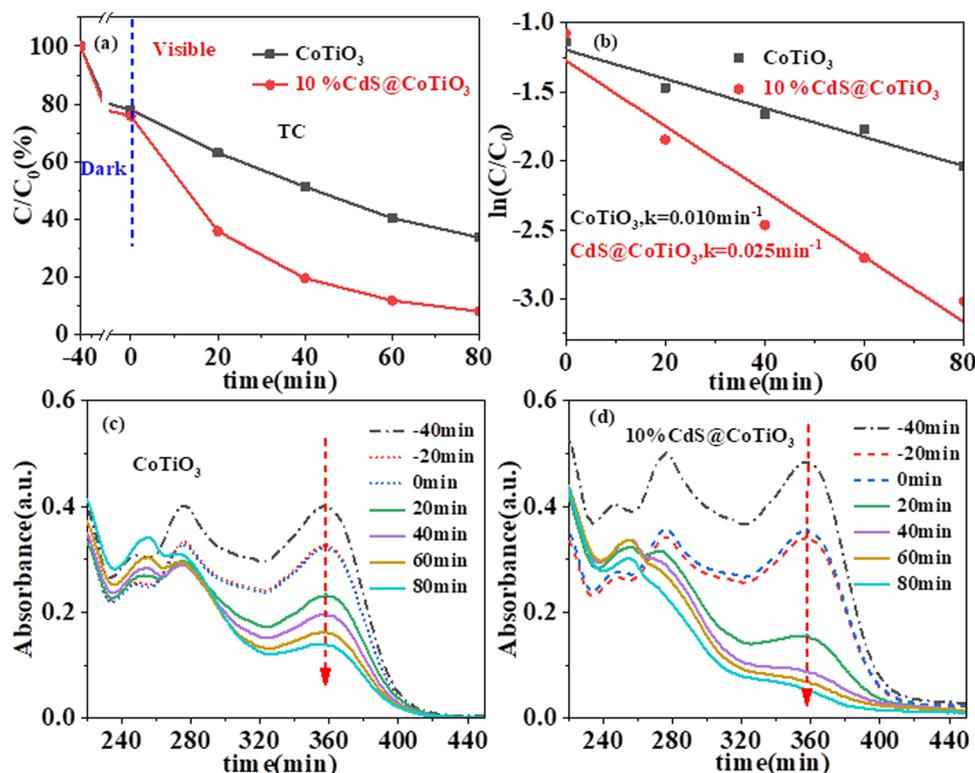


Fig. 7 (a) Photocatalytic degradation of TC aqueous solutions, and the (b) corresponding kinetic fitting curves over pure  $\text{CoTiO}_3$  and (d) 10%  $\text{CdS@CoTiO}_3$ ; time-dependent absorption spectra of TC degradation: (c) pure  $\text{CoTiO}_3$ , (d) 10%  $\text{CdS@CoTiO}_3$ .

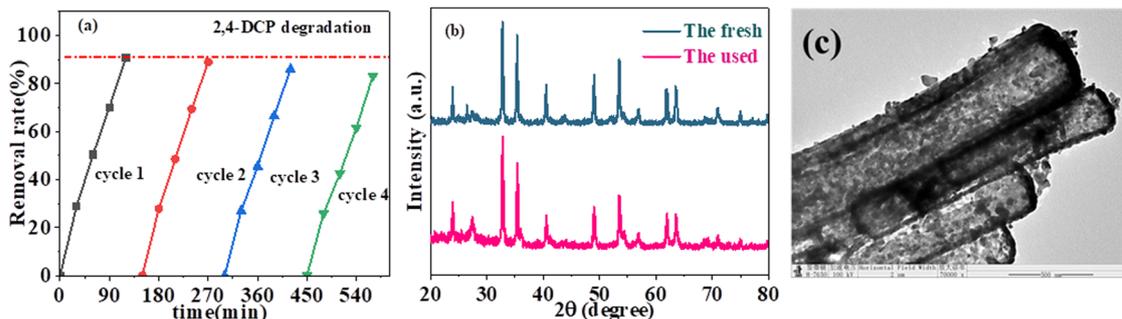


Fig. 8 (a) Reusability of the catalyst for the removal of 2,4-DCP under visible-light irradiation, (b) XRD patterns before and after the photocatalytic tests, and (c) TEM image of the CdS@CoTiO<sub>3</sub> hybrid after use.

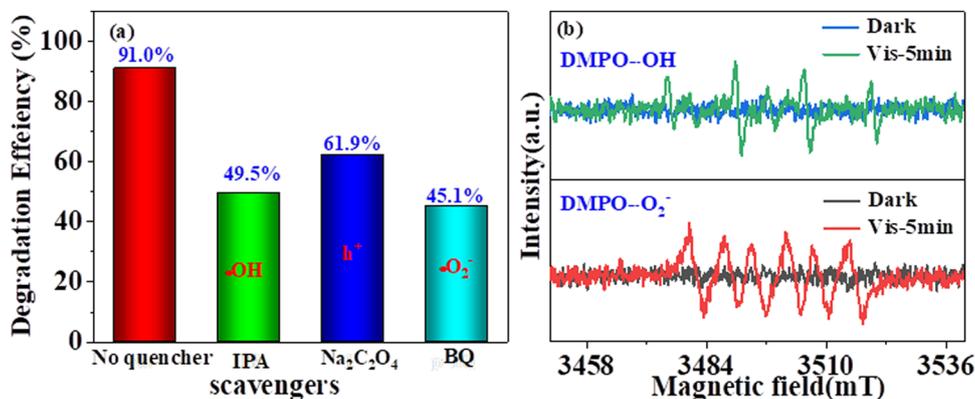
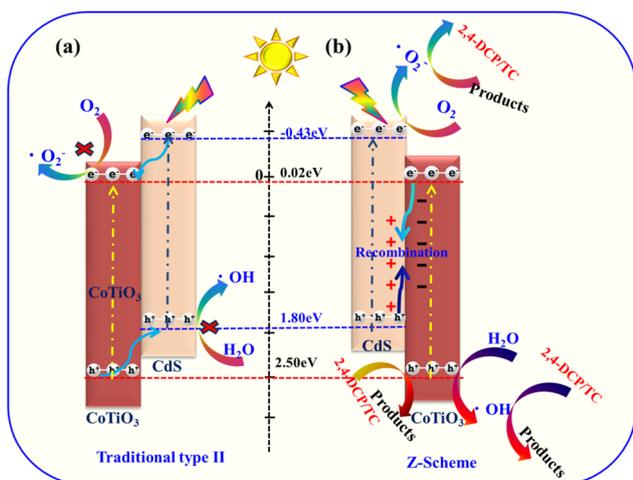


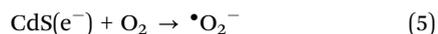
Fig. 9 (a) Influence of various scavengers on the degradation of 2,4-DCP and (b) EPR spectra in the dark and under visible-light irradiation for DMPO-•OH and DMPO-•O<sub>2</sub><sup>-</sup>.

of 1 : 1 : 1 : 1) were also detected in the CdS@CoTiO<sub>3</sub> photocatalytic degradation system, illustrating the formation of •O<sub>2</sub><sup>-</sup>. The EPR results unambiguously suggested that •OH and •O<sub>2</sub><sup>-</sup> radicals were both formed over the CdS@CoTiO<sub>3</sub> hybrids, which was well consistent with the active species trapping experimental results.

Based on the energy band positions of CdS and CoTiO<sub>3</sub>, as well as the results from the radical trapping tests and EPR analysis, a typical Z-scheme mechanism over the core-shell structured CdS@CoTiO<sub>3</sub> is proposed, as shown in Scheme 2. In this mechanism, owing to the narrow band gaps, both CdS and CoTiO<sub>3</sub> can be excited by visible light to generate electrons (e<sup>-</sup>) in the conduction band (CB) and positive holes (h<sup>+</sup>) in the valence band (VB) simultaneously. Generally, e<sup>-</sup> transfer occurs from the E<sub>C,CdS</sub> to the E<sub>C,CoTiO<sub>3</sub></sub>, meanwhile h<sup>+</sup> accumulate at E<sub>V,CdS</sub> (Scheme 2a, type-II heterojunctions). In such a case of charge-transfer behavior, the E<sub>V,CdS</sub> is not sufficient to oxidize OH<sup>-</sup> (E<sub>OH<sup>-</sup>→•OH</sub> = 2.38 eV vs. NHE) and the E<sub>C,CoTiO<sub>3</sub></sub> is unable to reduce O<sub>2</sub> to generate •O<sub>2</sub><sup>-</sup> (E<sub>O<sub>2</sub>→•O<sub>2</sub><sup>-</sup></sub> = 0.33 eV vs. NHE).<sup>61,62</sup> Clearly, this is not consistent with the results of the trapping experiments, where •O<sub>2</sub><sup>-</sup> and •OH were found to be the major active species in the photocatalytic degradation 2,4-DCP system. For the Z-scheme mechanism, the photoinduced e<sup>-</sup> in the CB of CoTiO<sub>3</sub> can transfer to the VB of CdS and recombine with the photogenerated h<sup>+</sup>, which not only inhibits the self-recombination of e<sup>-</sup>/h<sup>+</sup> pairs in both CdS and CoTiO<sub>3</sub> but also preserves the more negative CB potential of CdS and a more positive VB potential of CoTiO<sub>3</sub> for the photocatalytic reaction, thereby resulting in an efficient 2,4-DCP mineralization reaction. Drawing from the above analysis, the possible reaction equations could be elucidated as follows in eqn (4)–(7):



Scheme 2 Illustration of the photocatalytic degradation mechanisms for 3D CdS@CoTiO<sub>3</sub> heterojunctions: (a) type II and (b) Z-scheme.



From the architectural point of view, the hierarchical hollow core-shell nature of the CdS@CoTiO<sub>3</sub> photocatalyst with large internal cavities indeed endow it with sufficient active sites, while also shortening the transport distance of photoinduced charges as well as improving the capture capacity for visible light, simultaneously.

## 4. Conclusions

In summary, a hierarchical core-shell CdS@CoTiO<sub>3</sub> photocatalyst was constructed by uniformly modifying a CdS NPs layer on the surface of CoTiO<sub>3</sub> hollow NRs. The experimental investigation and EPR analytical results confirmed the existence of a Z-scheme photoinduced carrier transport path between CdS and CoTiO<sub>3</sub>. The Z-scheme heterojunction configuration combining this delicate architecture greatly decreased the recombination of photogenerated e<sup>-</sup>/h<sup>+</sup> pairs while still retaining a strong oxidation and reduction ability, resulting in a dramatically improved photocatalytic degradation activity for 2,4-DCP and TC. Additionally, the hollow core-shell nanorod structure could maintain good photocatalytic stability. It is speculated that the strategy adopted in this work could also be expanded to other hollow CoTiO<sub>3</sub>-based materials for applications not only in photocatalysis but also in other fields, such as energy storage and conversion, and electrochemistry.

## Conflicts of interest

There are no conflicts to declare.

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