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Doping SnO₂ with metal ions of varying valence states: discerning the importance of active surface oxygen species vs. acid sites for C₃H₈ and CO oxidation†

Haiming Yan, Teng Liu, Yu Lv, Xianglan Xu,  Junwei Xu, Xiuzhong Fang and Xiang Wang *

To elucidate the valence state effect of doping cations, Li⁺, Mg²⁺, Cr³⁺, Zr⁴⁺ and Nb⁵⁺ with radii similar to Sn⁴⁺ (CN = 6) were chosen to dope tetragonal SnO₂. Cr³⁺, Zr⁴⁺ and Nb⁵⁺ can enter the SnO₂ lattice to produce solid solutions, thus creating more surface defects. However, Li⁺ and Mg²⁺ can only stay on the SnO₂ surface as nitrates, thus suppressing the surface defects. The rich surface defects facilitate the generation of active O₂[−]/O^{δ−} and acid sites on the solid solution catalysts, hence improving the reactivity. On the solid solution catalysts active for propane combustion, several reactive intermediates can be formed, but are negligible on those with low activity. It is confirmed that for propane combustion, surface acid sites play a more vital role than active oxygen sites. Nevertheless, for CO oxidation, the active oxygen sites play a more vital role than the acid sites.

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1. Introduction

Rapid global industrialization has led to the release of a large amount of volatile organic compounds (VOCs) into the atmosphere, which poses not only environmental problems but also various threats to human health.^{1–4} Therefore, the abatement of VOCs is one of the compelling global challenges faced by every country. Catalytic combustion stands out as one of the most promising methods for effectively eliminating these pollutants.^{5–9} Catalysts containing noble metals, such as Pt, Pd, Ru *etc.*, exhibit high activity for VOC combustion.^{10–14} However, the widespread application of noble metals has been restricted due to their high cost.^{5,15,16} The development of non-noble metal catalysts has considerable advantages because of their cost-effectiveness and greater availability of resources, which has received considerable attention from researchers in recent decades.^{17–20}

Due to its abundant lattice and surface defects/vacancies, tetragonal rutile SnO₂, as an n-type semiconducting oxide, finds applications in gas sensors, solar cells, semiconductor devices, and electrode materials.^{21–23} Previous investigations have validated that the surface defect enables the adsorption

and activation of gaseous O₂, thus resulting in the generation of abundant active O₂[−] and O₂^{2−} sites,^{24–26} whereas these surface oxygen species on pure SnO₂ can be relinquished when calcined at temperatures exceeding 300 °C, which thus limits its redox property and reactivity.^{27,28} The incorporation of secondary metal ions in the SnO₂ matrix has been reported to effectively stabilize the surface-active oxygen sites. For example, former studies have demonstrated that the introduction of Cu²⁺, Fe³⁺, Cr³⁺, Ta⁵⁺, Ce⁴⁺ and Nb⁵⁺ in the SnO₂ lattice to form solid solutions not only improves the concentration of active surface oxygen sites but also stabilizes them at temperatures higher than 300 °C, due to the formation of more lattice/surface defects and charge imbalance.^{28–33}

The utilization of solid solutions with diverse chemical compositions is crucial for a wide range of catalytic reactions, and has gained significant attention.^{16,34} As indicated by previous investigations, the formation of steady solid solutions can only be achieved if the two cations possess similar radii and electronegativities.^{35,36} The additive materials alter the bulk structure and surface defects of the host matrix, which depends on their ionic radii and valence states.

Nevertheless, the effect of doping ions with varied valence states on the structure and reactivity of solid solution catalysts has not yet been investigated systematically. Researchers still lack clear understanding of some important fundamental scientific issues. Aiming to resolve this problem, Li⁺, Mg²⁺, Cr³⁺, Zr⁴⁺ and Nb⁵⁺ cations with different valence states but having radii similar to Sn⁴⁺ have been selected to modify SnO₂

Key Laboratory of Jiangxi Province for Environment and Energy Catalysis, School of Chemistry and Chemical Engineering, Nanchang University, Nanchang, Jiangxi, 330031, P.R. China. E-mail: xwang23@ncu.edu.cn

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by using a sol-gel method, with the expectation to incorporate them into the lattice of SnO_2 to obtain solid solutions. The prepared catalysts are tested *via* CO and propane oxidation to elucidate their surface-active sites and reactivity from different angles. Indeed, it was discovered that while Cr^{3+} , Zr^{4+} and Nb^{5+} can be successfully doped in the SnO_2 matrix to achieve solid solutions, Li^+ and Mg^{2+} cannot enter the SnO_2 lattice because the valence state deviation is too large. Moreover, we found that for CO oxidation, the active surface oxygen sites play a more critical role than the surface acid sites. However, for propane oxidation, the surface acid sites play a more crucial role than the active surface oxygen sites.

2. Experimental section

2.1 Catalyst preparation

The samples were fabricated using a sol-gel method. The detailed preparation process is outlined in the ESI.†

2.2 Catalyst characterization and activity testing

The samples were characterized by X-ray diffraction (XRD), Raman Spectroscopy, X-ray photoelectron spectroscopy (XPS),

H_2 temperature programmed reduction (H_2 -TPR), O_2 temperature programmed desorption (O_2 -TPD), NH_3 temperature programmed desorption (NH_3 -TPD), propane temperature programmed desorption (propane-TPD), electron paramagnetic resonance (EPR), high resolution transmission electron microscope (HRTEM), scanning transmission electron microscopy-mapping (STEM-mapping) and *in situ* DRIFTS techniques. The catalytic performance was assessed through CO oxidation and propane combustion. Notably, to get reliable and useful reaction kinetic data to guide catalyst design, the reaction feeds simulate real after-treatment processes by using air as an oxidant, which consists of $\sim 78\%$ N_2 and 21% O_2 . The equipment models, experimental procedures, reaction conditions and operation parameters are described in the ESI.†

3. Results and discussion

3.1 XRD and Raman analysis

The XRD patterns are manifested in Fig. 1(a) for the catalysts, and the quantification results are listed in Table 1. Three strongest peaks around 26.3° (110), 33.6° (101) and 51.5° (211), which are characteristic of tetragonal rutile SnO_2 , are observed for all the samples,²⁴ but with varied intensity. The lattice parameters of the samples, as presented in Table 1, exhibit a remarkable resemblance to those of the individual SnO_2 , thereby substantiating the retention of the tetragonal rutile structure. The absence of any peak associated with the dopant cations suggests their possible incorporation into the SnO_2 lattice to produce solid solutions, or their high dispersion on the surface of SnO_2 , thereby evading detection *via* XRD.

Theoretically, to effectively establish a solid solution between two metal oxides, the two metal cations must have a similar radius and electronegativity.^{31,35} The Sn^{4+} cations in rutile SnO_2 have a coordination number (CN) of 6 and possess an ionic radius of 0.69 \AA . The XPS analysis in this study has confirmed the predominant presence of the doping metal cations such as Li^+ , Mg^{2+} , Cr^{3+} , Zr^{4+} and Nb^{5+} . Further details of these findings will be described in the XPS section. If these cations possess a CN of 6, their radii will be 0.76 , 0.72 , 0.62 , 0.72 and 0.64 \AA , respectively, which are similar to that of Sn^{4+} . According to the basic requirement, it is theoretically feasible to generate stable solid solutions between SnO_2 and these cations, whereas the valence states of all cations differ from Sn^{4+} except for Zr^{4+} , thus potentially affecting the generation of a solid solution structure.

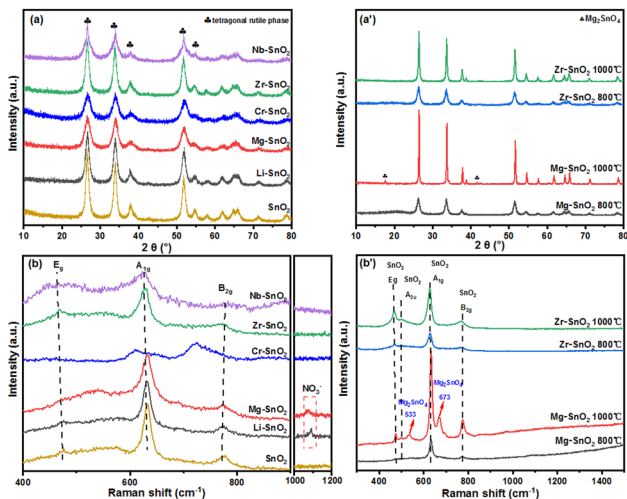


Fig. 1 Structural identification of the catalysts. (a) XRD patterns of the catalysts calcined at 500°C ; (a') XRD patterns of Mg-SnO_2 and Zr-SnO_2 calcined at 800 and 1000°C ; (b) Raman spectra of the catalysts calcined at 500°C ; (b') Raman spectra of Mg-SnO_2 and Zr-SnO_2 calcined at 800 and 1000°C .

Table 1 The physicochemical properties

Catalysts	Coordination number (CN)	Dopant ionic radius (\AA)	Sn/M molar ratio ^a	Lattice parameters			Average crystallite size ^b (nm)	Average grain size ^c	Surface area ($\text{m}^2 \text{g}^{-1}$)
				$a = b$ (\AA)	c (\AA)	$\alpha = \beta = \gamma$ ($^\circ$)			
SnO_2	6	0.69	—	4.806	3.211	90	7.2	—	24.2
Li-SnO_2	6	0.76	8.0/1	4.778	3.225	90	6.9	—	26.8
Mg-SnO_2	6	0.72	7.0/1	4.746	3.210	90	4.8	4.9	42.6
Cr-SnO_2	6	0.62	8.9/1	4.767	3.201	90	5.4	5.8	40.5
Zr-SnO_2	6	0.72	9.0/1	4.810	3.212	90	6.1	6.2	30.7
Nb-SnO_2	6	0.64	9.0/1	4.796	3.206	90	5.8	6.1	32.3

^a Measured *via* ICP. ^b Calculated using the Scherrer equation with the XRD (110) peak of SnO_2 . ^c Measured *via* HRTEM.

Hence, the catalysts were subjected to Raman analysis for further clarification. The Raman spectra of pure SnO_2 , as illustrated in Fig. 1(b), exhibit an E_g peak at 476 cm^{-1} , a prominent A_{1g} peak at 633 cm^{-1} , and a minor B_{2g} peak at 775 cm^{-1} .²⁵ Analogously, all the modified catalysts also show these three typical Raman bands, confirming that SnO_2 could still be in the crystalline phase. However, the E_g , A_{1g} and B_{2g} peaks of Cr^{3+} , Zr^{4+} and Nb^{5+} -doped samples are shifted and broadened in contrast to those of pure SnO_2 , which may be attributed to variations in the SnO_2 lattice parameters and the presence of surface vacancies/defects resulting from a solid solution structure.³⁰ In contrast, the samples doped by Li^+ and Mg^{2+} show spectra similar to that of individual SnO_2 without evident peak shift and broadening, and a characteristic peak of nitrate was observed around 1075 cm^{-1} .³⁷ This suggests the presence of Li^+ and Mg^{2+} on the surface of SnO_2 as nitrates rather than their incorporation into its matrix to produce a solid solution structure. Although Li^+ and Mg^{2+} have radii similar to that of Sn^{4+} , their valence states are very different from that of Sn^{4+} , which may result in too big difference in electronegativity to generate a solid solution structure.

To validate this further, Mg-SnO_2 and Zr-SnO_2 , the two typical samples, were calcined at higher temperatures of 800 and 1000 °C, and analysed *via* XRD and Raman spectroscopy. As depicted in Fig. 1(a'), the tetragonal rutile SnO_2 phase remains the sole crystalline phase for Zr-SnO_2 , providing evidence for the generation of a stable solid solution structure, whereas for the Mg-SnO_2 sample, when being calcined at 1000 °C, a new Mg_2SnO_4 phase appears, accompanying the disappearance of the surface $\text{Mg}(\text{NO}_3)_2$ species, which is also confirmed by the Raman results shown in Fig. 1(b'). This testifies again that for the Mg-doped samples, $\text{Mg}(\text{NO}_3)_2$ is present on its surface rather than being incorporated into the matrix of SnO_2 . Without the restriction of the crystal lattice, $\text{Mg}(\text{NO}_3)_2$ reacted with SnO_2 at high temperature to produce the Mg_2SnO_4 compound.³⁸ Indeed, this has confirmed that Cr^{3+} , Zr^{4+} and Nb^{5+} are incorporated into the lattice of tetragonal SnO_2 to form solid solutions, while Mg^{2+} and Li^+ are present on the surface as nitrates.

All the modified samples exhibit larger surface areas and smaller crystallite sizes than pure SnO_2 (Table 1). The solid solution structure typically hinders the crystallization of materials due to disparities in radius and charge distribution.^{29,39} Therefore, it is not difficult to understand that Cr-SnO_2 , Zr-SnO_2 and Nb-SnO_2 have improved surface areas. For Li-SnO_2 and Mg-SnO_2 , although a solid solution structure is not formed, the dispersion of Li and Mg nitrates on the surface can also hinder the crystallization process, thereby resulting in improved surface areas.⁴⁰

3.2 TEM, HRTEM and STEM-mapping results

The morphologies, grain sizes, crystallite growth, and elemental distribution of some representative catalysts have been analysed using TEM and HRTEM and STEM-mapping techniques. In Fig. 2, the TEM images reveal that the samples consist of irregular spherical grains with average sizes comparable to their corresponding crystallite sizes, providing evidence that there is no

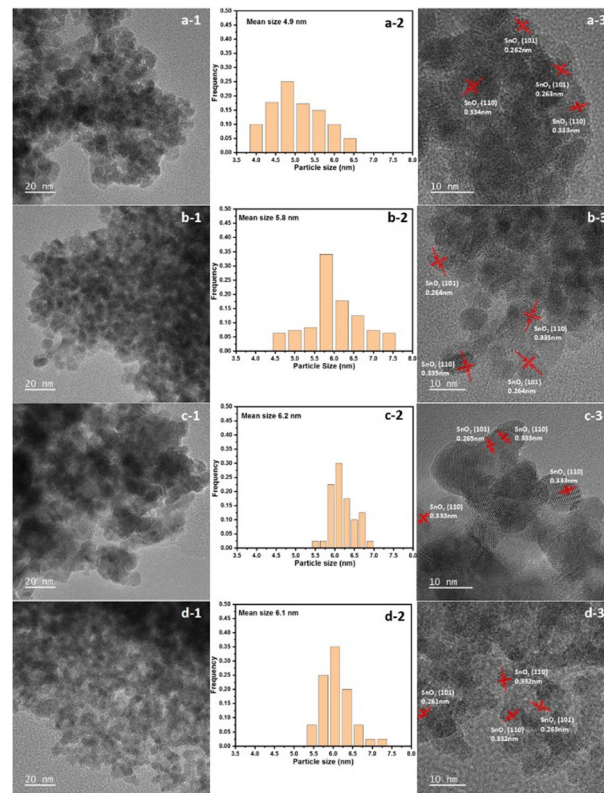


Fig. 2 HRTEM images and particle size distribution profiles: (a) Mg-SnO_2 ; (b) Cr-SnO_2 ; (c) Zr-SnO_2 ; (d) Nb-SnO_2 .

secondary agglomeration of the crystallites. The HRTEM images in Fig. 2 demonstrate that (110) and (101) diffraction planes of the SnO_2 phase were detected distinctly for all the samples, but no crystalline phases related to the dopant oxides can be detected. This proves that for the solid solution samples, the dopant cations are substituted into the matrix, and for the samples without a solid solution structure, the dopant cations are dispersed finely on the surface as nitrates, as testified by Raman results.

The STEM-mapping images in Fig. 3 reveal a nearly uniform distribution of Sn, M, and O across all the samples, further supporting the presence of Cr^{3+} , Zr^{4+} and Nb^{5+} in the SnO_2 matrix to generate stable solid solutions, but Mg^{2+} stays on the surface of SnO_2 in the form of highly dispersed nitrate.

3.3 Activity tests

To investigate the impact of different dopants on catalytic activity, we have conducted tests on the catalysts for CO and propane oxidation. Notably, CO_2 is the only product detected for both reactions in the tested temperature region. The individual SnO_2 exhibits certain activity, as shown in Fig. 4(a), due to the existence of active surface deficient oxygen and lattice oxygen species, on which CO is completely oxidized at 340 °C. It should be noted that Cr^{3+} and Nb^{5+} doping improves the activity, and the activity of Zr-SnO_2 is close to that of pure SnO_2 . In contrast, Li^+ and Mg^{2+} doping clearly degrades the activity, on which 100% CO conversion is achieved at 410 °C and 420 °C,

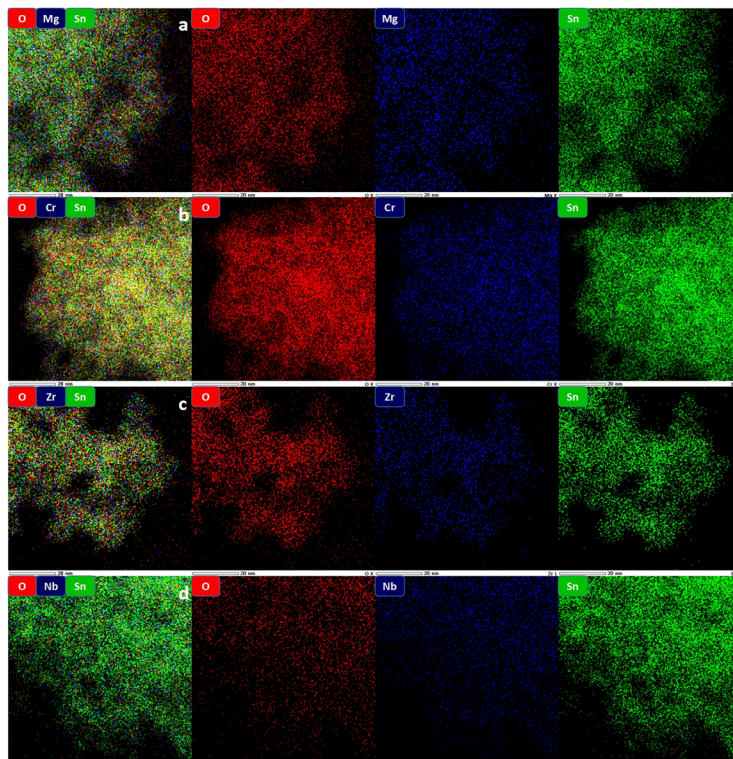


Fig. 3 STEM-mapping images of the catalysts: (a) Mg-SnO₂; (b) Cr-SnO₂; (c) Zr-SnO₂; (d) Nb-SnO₂.

respectively. Among all the catalysts, Nb-SnO₂ shows the highest activity, which can oxidize CO completely at 280 °C.

Arrhenius curves obtained under differential conditions are shown in Fig. 4(b). For clarity, Table 2 lists the CO conversion

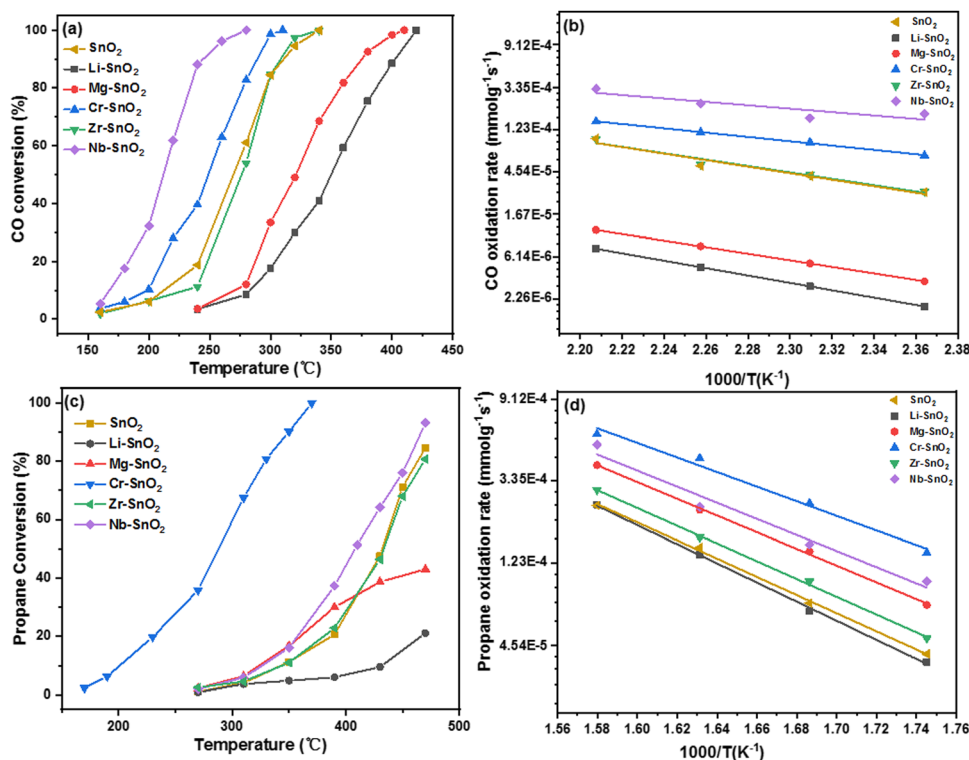


Fig. 4 CO and propane oxidation performance on the catalysts. (a) CO conversion; (b) Arrhenius plots for CO conversion; (c) propane conversion; (d) Arrhenius plots for propane conversion.

Table 2 CO and propane oxidation activity over M-SnO₂ catalysts

Samples	R_w^a (10^{-5} mmol g ⁻¹ s ⁻¹)		R_s^b (10^{-6} mmol s ⁻¹ m ⁻²)		E_a (kJ mol ⁻¹)	
	CO	Propane	CO	Propane	CO	Propane
SnO ₂	2.81	4.08	1.16	1.67	64.2	92.1
Li-SnO ₂	0.74	3.70	0.28	1.38	72.4	96.9
Mg-SnO ₂	1.15	7.43	0.27	1.59	64.2	84.7
Cr-SnO ₂	15.00	14.00	3.69	3.46	42.5	74.0
Zr-SnO ₂	2.86	4.93	0.93	1.66	62.8	90.0
Nb-SnO ₂	32.32	9.92	10.00	3.07	36.9	81.6

^a R_w , the differential rates normalized by catalyst weight. Measured at 180 °C for CO oxidation and at 300 °C for propane combustion. ^b R_s , the differential rates normalized by catalyst surface area. Measured at 180 °C for CO oxidation and at 300 °C for propane combustion.

rates measured at 180 °C, that is, R_w and R_s normalized by catalyst weight and surface area, and the apparent activation energies. The reaction rates display a comparable trend to the overall activity, which is sequenced as Nb-SnO₂ > Cr-SnO₂ > Zr-SnO₂ ≈ SnO₂ > Mg-SnO₂ > Li-SnO₂. Compared to pure SnO₂, Nb-SnO₂ and Cr-SnO₂ have higher reaction rates and lower activation energy, which suggests that the incorporation of Nb⁵⁺ and Cr³⁺ into the solid solution structure leads to the formation of a greater number of active surface sites for CO oxidation. Previous studies have demonstrated that active surface oxygen species play a key role in CO oxidation reactions.^{41–43} The formation of solid solutions leads to an increased abundance of surface vacancies, thereby promoting the creation of additional reactive oxygen sites on the surface and facilitating CO oxidation.^{44,45} However, the presence of Li and Mg nitrates on the SnO₂ surface is obviously harmful to the activity.

Interestingly, the results in Fig. 4(c) and (d) demonstrate that the sequence of both overall and intrinsic activity for propane deep oxidation differs from that observed in CO oxidation. The overall activity is sequenced as Cr-SnO₂ > Nb-SnO₂ > Zr-SnO₂ ≈ SnO₂ > Mg-SnO₂ > Li-SnO₂. Basically, the intrinsic activity judged by R_s obeys the same sequence. Cr-SnO₂ shows much higher activity than Nb-SnO₂ and other catalysts, which suggests that for propane deep oxidation, the surface acid sites might play a more vital role than the active

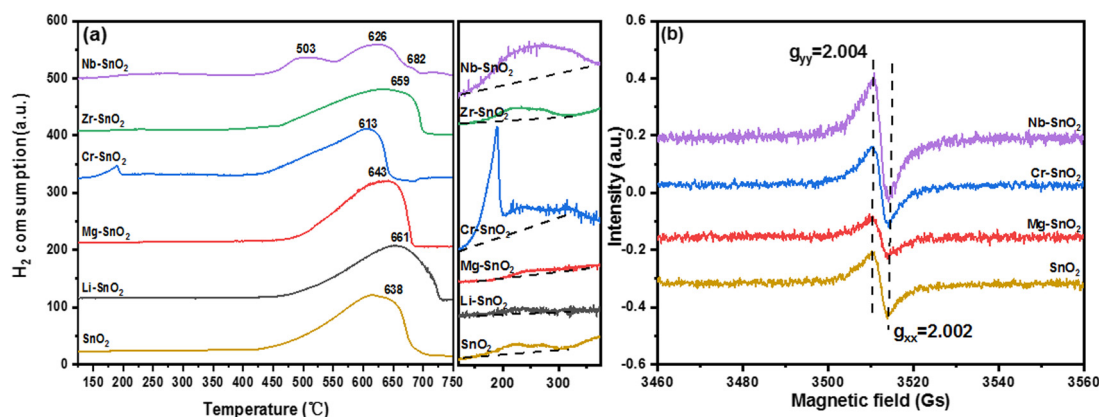
oxygen species. Therefore, it is proposed that surface acid sites could play a more critical role here, which can adsorb and activate those organic molecules with rich electrons, such as propane molecules in this work.^{46,47}

3.4 H₂-TPR, O₂-TPD and EPR studies

To examine the impact of introducing different valence state ions, the redox properties were tested by conducting H₂-TPR. As depicted in Fig. 5(a), the reduction peaks of all samples can be categorized into two distinct groups: one group ranging from 150 to 350 °C, and another group above 400 °C. The minor peak observed at 150–300 °C can be ascribed to the reduction of surface deficiency oxygen, which is typical for SnO₂ and related to its n-type semiconducting property,^{28,39} whereas this facile oxygen species will be depleted on individual SnO₂ when the treating temperature exceeds 400 °C.^{27,28} As manifested by the partly enlarged profiles beside the main figure and the quantification results in Table 3, the formation of the solid solution can significantly enhance the concentration of this active oxygen species and stabilize it at elevated temperature, especially for Cr-SnO₂ and Nb-SnO₂. In contrast, for Li-SnO₂ and Mg-SnO₂ with the formation of Li and Mg nitrates on the surface, the concentration of this deficiency oxygen species decreases evidently, in good accordance with the CO oxidation activity sequence. This has proven that the surface deficient oxygen plays a crucial role for CO oxidation, although it is also necessary for propane deep oxidation.

Table 3 Quantified H₂-TPR and O₂-TPD results

Catalysts	H ₂ uptake amount (mmol g ⁻¹)		O/Sn atomic ratio	O ₂ desorption amount (μmol g ⁻¹)		
	Below 350 °C	Above 350 °C		α	β	α + β
SnO ₂	0.14	13.27	2.0	0.62	36.86	37.48
Li-SnO ₂	0.06	12.67	2.1	0.37	31.83	32.20
Mg-SnO ₂	0.05	12.11	2.0	0.47	35.41	35.88
Cr-SnO ₂	0.63	11.13	1.9	2.66	36.84	39.50
Zr-SnO ₂	0.18	11.81	2.0	0.60	37.17	37.77
Nb-SnO ₂	0.80	11.62	2.0	3.08	38.65	41.73

Fig. 5 (a) H₂-TPR profiles of the catalysts; (b) EPR spectra of some typical catalysts.

EPR experiments were thus carried out using some typical samples to identify the facile oxygen species. The catalysts exhibit two bands at $g = 2.004$ and 2.002 , as depicted in Fig. 5(b), which are assignable to surface O_2^- .⁴⁸ The order of the peak integrated areas is as follows, $Nb-SnO_2 > Cr-SnO_2 > SnO_2 > Mg-SnO_2$, in agreement with the CO oxidation activity sequence. The existence of surface O_2^- suggests that it may serve as one of the significant oxygen centres for CO oxidation.

The broad and significant reduction peak above $400^\circ C$ can be attributed to the reduction of Sn^{4+} to Sn^0 ,^{24,25} which is further validated by the O/Sn molar ratio around 2 as shown in Table 3. With the doping of Nb^{5+} and Cr^{3+} to form solid solutions, the mobility of the lattice oxygen is obviously increased, thus becoming easier to be reduced.

The oxygen characteristics of the catalysts have been analysed from different perspectives using O_2 -TPD. The desorption peaks observed in Fig. S1 (ESI[†]) can be categorized into two groups: one group, denoted as α type, occurs in the temperature range of $50-200^\circ C$ and corresponds to the release of loosely bonded surface oxygen,^{24,49} while the other group, referred as β type, appears between 200 and $700^\circ C$ and is associated with the release of surface lattice O^{2-} .^{24,49} As quantified in Table 3, the integrated areas of the α peak for $Cr-SnO_2$ and $Nb-SnO_2$ show significant improvement, indicating the presence of a larger amount of surface facile oxygen compared to individual SnO_2 . For $Li-SnO_2$ and $Mg-SnO_2$, the amount of this part of oxygen species becomes smaller. The variation in the integrated areas of the β peak exhibits a comparable trend, indicating that the surface lattice oxygen of $Cr-SnO_2$ and $Nb-SnO_2$ becomes also more mobile. However, since most of the surface lattice oxygen releases at much higher temperature, it is believed that the surface facile oxygen could contribute predominantly to the reactions. The amount of the desorbed facile oxygen is sequenced as $Nb-SnO_2 > Cr-SnO_2 > SnO_2 \approx Zr-SnO_2 > Mg-SnO_2 > Li-SnO_2$, being consistent with the CO oxidation performance.

3.5 *In situ* DRIFTS investigation of the active surface oxygen sites

To further distinguish the surface-active oxygen types and their reactivity, the representative $Mg-SnO_2$ and $Cr-SnO_2$ catalysts were studied by *in situ* DRIFTS. The results obtained on the $Cr-SnO_2$ catalyst are described in detail here. The sample was initially subjected to a cleaning process at $500^\circ C$ in an Ar flow (30 mL min^{-1}) for 60 minutes to ensure surface cleanliness and eliminate the presence of chemisorbed active oxygen that had formed during the initial stages. Subsequently, it was cooled down to $300^\circ C$. The *in situ* cell was then supplied with a stream of 10% O_2/Ar . Fig. 6(a) displays that compared with the clean catalyst (0 min line), two groups of IR peaks were observed immediately after 1 minute, with one at $980-1020\text{ cm}^{-1}$ and the other around 1260 cm^{-1} . The former is assigned to the surface superoxide anions (O_2^-), and the latter is ascribed to the surface oxygen species defined as $O^{\delta-}$ ($0 < \delta < 1$).^{49,50} After 5 minutes, the two peaks reached a steady state, proving that the surface was fully oxidized by O_2 . After a duration of

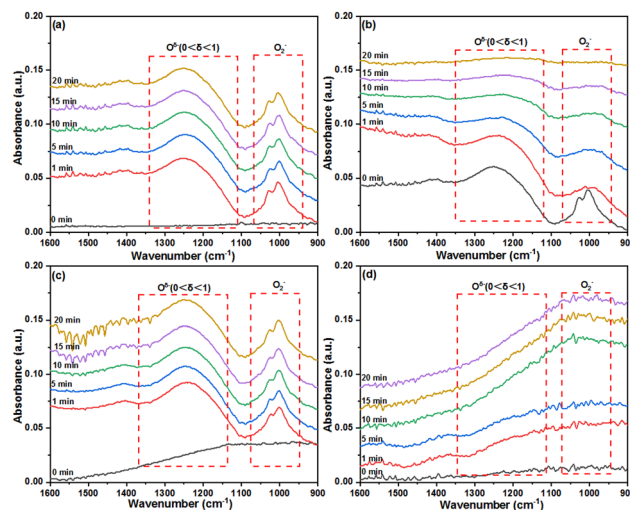


Fig. 6 *In situ* DRIFTS spectra of $Cr-SnO_2$ obtained at $300^\circ C$ (a) treated in a 10% O_2/Ar flow, (b) treated in a 10% H_2/Ar flow and (c) switched back to the 10% O_2/Ar flow. (d) *In situ* DRIFTS spectra of $Mg-SnO_2$ treated in a 10% O_2/Ar flow at $300^\circ C$.

20 minutes, a feed consisting of 10% H_2/Ar was introduced into the cell while maintaining the same temperature. The results depicted in Fig. 6(b) demonstrate a fast and complete depletion of both types of surface oxygen anions within 10 minutes, indicating the active and reducible nature of these two oxygen species. After this step, the flow of 10% O_2/Ar was redirected into the cell while maintaining the same temperature. As shown in Fig. 6(c), the two types of surface oxygen can be fully restored after 10 minutes, even if they are completely consumed by hydrogen.

In contrast, the *in situ* DRIFTS results in Fig. 6(d) obtained with $Mg-SnO_2$ exhibit that no any surface $O^{\delta-}$ ($0 < \delta < 1$) is present, and the generation of surface O_2^- species is also negligible, which explains its poor oxidation activity. In brief, the *in situ* DRIFTS results have substantiated that besides surface O_2^- species, $O^{\delta-}$ ($0 < \delta < 1$) might be another kind of facile oxygen sites contributing to the reactions for those samples possessing a solid solution structure.

3.6 XPS analysis

The surface properties have been analysed using X-ray photoelectron spectroscopy (XPS). Fig. S2 (ESI[†]) presents that two symmetrical peaks were observed around 487.1 and 495.5 eV for all the catalysts, corresponding to $Sn\ 3d_{5/2}$ and $Sn\ 3d_{3/2}$ of Sn^{4+} , respectively.⁵¹ The presence of different metal cations is observed to cause a noticeable shift towards lower binding energy regions for the two peaks, indicating that Sn gains electrons from the dopants. The binding-energy gap (ΔE) between the two peaks has been calculated and presented in Table S1 (ESI[†]). In contrast to pure SnO_2 , all the samples exhibit a slight decrease in ΔE , indicating that the chemical environment of lattice Sn^{4+} is modified upon doping with metal ions either within the lattice or on the surface.³¹ As shown in Fig. S3

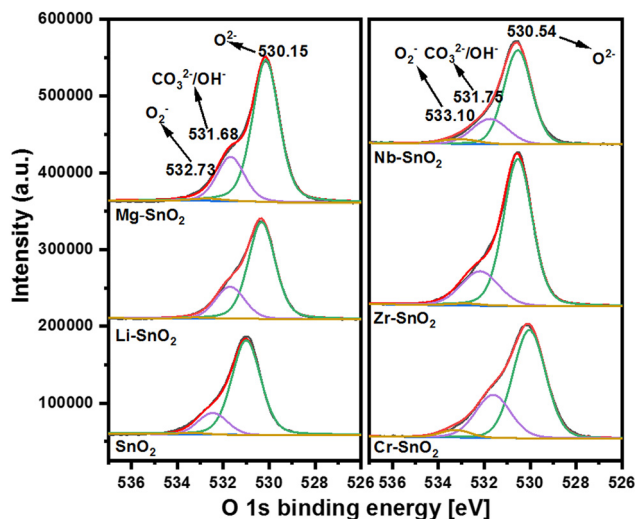


Fig. 7 XPS O 1s spectra of M-SnO₂ catalysts.

(ESI[†]), it has been confirmed that the doping cations are mainly in the states of Li⁺, Mg²⁺, Cr³⁺, Zr⁴⁺ and Nb⁵⁺, referring to the standard binding energies of the corresponding metal oxides.

The XPS O 1s peaks are thoroughly analysed based on the *in situ* DRIFTS and EPR results. The unsymmetrical O 1s peak, as depicted in Fig. 7, can be fitted into three peaks centered at approximately 530.0, 531.0, and 533.0 eV, corresponding to the surface lattice O²⁻, CO₃²⁻/OH⁻ and O₂⁻ anchored on the defects of the surface.^{52,53} The O₂⁻/O²⁻ ratios have been calculated in Table S1 (ESI[†]) to discern the effect of different dopants on the quantity of surface-active oxygen. Interestingly, the ratios are sequenced as Nb-SnO₂ > Cr-SnO₂ > Zr-SnO₂ ≈ SnO₂ > Mg-SnO₂ > Li-SnO₂, well consistent with the O₂-TPD and CO oxidation activity order. Nb-SnO₂ and Cr-SnO₂, the two samples with a solid solution structure, show much higher O₂⁻/O²⁻ ratios than pure SnO₂. In contrast, the doping of Zr⁴⁺ into the SnO₂ lattice improves the O₂⁻/O²⁻ ratio only slightly. This seems to imply that doping the SnO₂ lattice with cations having varied valence states (Cr³⁺ and Nb⁵⁺) can create more surface defects than doping with cations having the same valence state (Zr⁴⁺), thus inducing the creation of a larger amount of surface-active oxygen anions. Notably, Li-SnO₂ and Mg-SnO₂ possess lower O₂⁻/O²⁻ ratios than pure SnO₂, testifying that the Li and Mg nitrates dispersed finely on the surface can suppress surface defects. The observed O₂⁻/O²⁻ ratio change achieved *via* XPS is in agreement with the H₂-TPR, O₂-TPD, and EPR results, demonstrating consistency with the catalytic activity, especially for CO oxidation.

3.7 NH₃-TPD and propane-TPD studies

As discussed above, for propane deep oxidation, surface acid sites might play a more important role than surface-active oxygen centres. It reported previously that the surface acid sites are very important for the adsorption and activation of the toluene molecule, owing to the delocalized π^*_o bond with rich electrons in its structure, thereby significantly improving the combustion activity.^{24,25,54} Therefore, the surface acidity was

Table 4 NH₃-TPD and propane-TPD quantification results

Sample	NH ₃ desorption amount ($\times 10^{-2}$ mmol g ⁻¹)	Relative propane desorption amount (a.u.)
SnO ₂	0.74	25.27
Li-SnO ₂	2.01	28.37
Mg-SnO ₂	2.61	30.85
Cr-SnO ₂	5.20	100
Zr-SnO ₂	1.09	23.95
Nb-SnO ₂	3.67	65.87

investigated by conducting NH₃-TPD experiments. As demonstrated in Fig. S4(a) (ESI[†]), the presence of surface acidic sites is confirmed by the wide NH₃ release peak observed in the range of 50–300 °C for all the catalysts. The calculation results in Table 4 show that the amount of surface acidic sites is ranked as follows: Cr-SnO₂ > Nb-SnO₂ > Mg-SnO₂ > Li-SnO₂ > Zr-SnO₂ > SnO₂, indicating that doping the metal ions into the SnO₂ lattice to produce solid solutions can generate higher surface acidity. The change trend of the acid site amount is more similar to the propane deep oxidation activity sequence, which seems to indicate that surface acid sites are more important than surface-active oxygen centres for propane oxidation.

The quantification results in Table 4 demonstrate that a solid solution sample with more surface defects generally owns a larger amount of surface acid sites. The previous reports indicated a strong correlation between the quantity of surface Lewis acid sites and surface defects.^{29,55,56} Fig. S4(b) (ESI[†]) displays the propane-TPD profiles, and the quantification results in Table 4 prove that the desorption amount of propane follows the same trend, which indicates that the two types of molecules share the same kind of adsorption sites. The relationship between the desorption amount of propane and the number of surface acidic sites is clearly positively correlated, as demonstrated in Fig. 8. Therefore, surface acidity is considered crucial for propane deep oxidation.

3.8 The competition of the adsorption sites between NH₃ and propane/CO probed by *in situ* DRIFTS

As supplementary studies, the adsorption of NH₃ and propane on Cr-SnO₂, the most active catalyst for propane combustion,

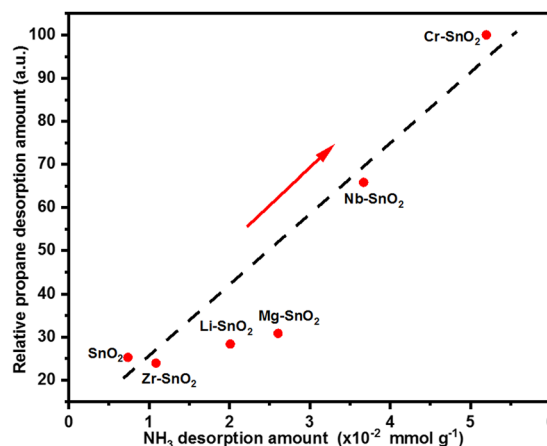


Fig. 8 Propane desorbing amount vs. acidic site amount.

was investigated using *in situ* DRIFTS results as well. The sample was first cleansed at 500 °C in an Ar flow (30 mL min⁻¹) for 60 minutes and then cooled down to 50 °C. Fig. 9(a-I) shows that after introducing NH₃, the range of N-H deformation modes below 1700 cm⁻¹ exhibited four distinct bands, which can be attributed to the adsorption of NH₃ on various acid sites. The bands at 1622 and 1271 cm⁻¹ are attributed to NH₃ bound to Lewis-acid sites.^{57,58} The bands observed at 1674 and 1450 cm⁻¹ can be assigned to the presence of NH₃ ions resulting from the protonation of NH₃ on Brønsted acid sites.^{57,58} Subsequently, a propane/Ar feed was introduced, and Fig. 9(a-II) does not show any change in the NH₃ adsorbing peaks. This suggests that the propane molecules cannot be adsorbed after the surface acid sites have been occupied by NH₃ molecules. In other words, both the NH₃ and propane molecules could share the same surface acid sites, as testified by the three kinds of TPD experiments in the last section.

To confirm this further, *in situ* DRIFTS experiments by adsorbing NH₃ and propane with a reversed order were also carried out. As shown in Fig. 9(b), after adsorbing propane at 50 °C firstly on the cleansed catalyst, three peaks are observed around 1617, 1454, and 1241 cm⁻¹. As labelled in the curves, all of the peaks belong to the dissociative adsorption of propane.^{59–61} In addition, a gaseous propane vibration peak is detected around 2968 cm⁻¹.⁶² After introducing NH₃, the

three characteristic peaks were replaced by the four typical peaks of adsorbing NH₃. Therefore, it is strongly proved that both NH₃ and propane are adsorbed on the acidic sites, which are crucial to activate propane molecules.

In addition, CO and NH₃ adsorption was also investigated using the *in situ* DRIFTS technique following the same procedures. Basically, a similar phenomenon is observed in the adsorption of NH₃ and propane. As demonstrated in Fig. 9(c), after the surface of Cr-SnO₂ is pre-saturated by NH₃ molecules, it is difficult to adsorb CO molecules. As shown in Fig. 9(d), after the surface of Cr-SnO₂ is pre-saturated by CO molecules again, NH₃ molecules can drive out the pre-adsorbed CO molecules. Since a CO molecule contains π -back bonding, surface acid sites are necessary for its adsorption/activation. But the surface-active oxygen sites may play a more critical role than the surface acid sites for CO oxidation, as discussed above.

3.9 The reaction intermediates probed by *in situ* DRIFTS

The possible pathway of propane oxidation was investigated through a series of *in situ* DRIFTS experiments carried out on Cr-SnO₂, recognized as the most active catalyst for this reaction in this work.

The first series: the sample was initially subjected to a purification process at 500 °C in an Ar flow (30 mL min⁻¹) for 60 minutes, followed by a decrease in temperature to 50 °C.

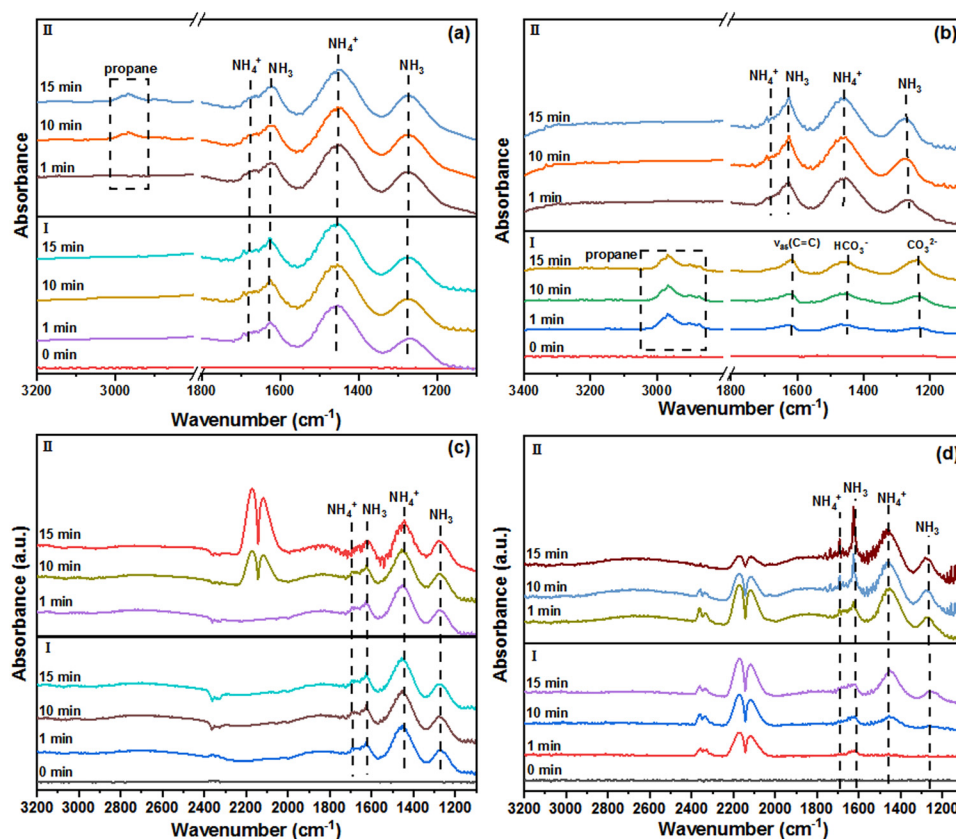


Fig. 9 *In situ* DRIFT adsorption spectra of Cr-SnO₂ at 50 °C, (a-I) treated in a NH₃/Ar stream, (a-II) treated in a propane/Ar stream, (b-I) treated in a propane/Ar stream, (b-II) treated in a NH₃/Ar stream, (c-I) treated in a NH₃/Ar stream, (c-II) treated in a CO/Ar stream, (d-I) treated in a CO/Ar stream, (d-II) treated in a NH₃/Ar stream.

Subsequently, a propane feed mixed with Ar was directed into the cell in the temperature range of 50–300 °C. As shown in Fig. 10(a), at 50 and 100 °C, the surface adsorption of propane is very weak, and gas phase propane is obviously detected at 2968 and 2982 cm⁻¹. However, after the temperature is increased to 200 °C, the initially very weak adsorption peaks can be evidently detected. The peaks observed at 2902 and 2876 cm⁻¹ can be attributed to the C-H vibrations of CH, CH₂, or CH₃ species, suggesting that propane undergoes adsorption, partial oxidation, and cracking on the catalyst surface.⁶³ At 50 °C, the peak at 1620 cm⁻¹ is assigned to the vibration of the C=C bond. When the temperature was increased to 100 °C, it was oxidized to form C=O bonds detected at 1676 cm⁻¹.⁶⁴ When the temperature is higher than 200 °C, new bands appeared at 1542, 1384, 1351, and 1424 cm⁻¹, which are assigned to ν_{as} (COO), δ_{s} (CH₃), O–C–O bonds in carboxylate species and carbonate species, respectively.^{64,65} The evidence exhibits that the presence of surface O²⁻ species facilitates the partial oxidation of propane adsorbed on the catalyst. As shown in Fig. S5(a) (ESI[†]), compared with Cr–SnO₂, the formation of those intermediate species is not observed on Mg–SnO₂, showing that propane cannot be sufficiently activated under the same condition. However, Fig. S5(b) (ESI[†]) indicates that much weaker intermediate peaks can still be detected on Nb–SnO₂, which indicates that propane can only be mildly activated.

The second series: as displayed in Fig. 10(b-I), full adsorption of propane was carried out at 300 °C to generate the intermediates detected in the first series of experiments. After this, 10% O₂/Ar was introduced at the same temperature. As shown in Fig. 10(b-II), the intermediate bands exhibited a quick decrease, particularly those in relation to the oxygenates. This observation provides evidence that these adsorption species may serve as reactive intermediates during propane combustion.

The third series: a batch of fresh catalysts was subjected to the same pretreatment process as in the first series of experiments for surface cleansing, followed by temperature controlling at 300 °C. After being fully oxidized in a 10% O₂/Ar feed, Fig. 10(c-I) shows that two groups of peaks around 1000 and 1200 cm⁻¹ appear, corresponding to O₂⁻ and O^{δ-} in a sequence. Afterwards, a propane/Ar stream was switched into the cell. Fig. 10(c-II) shows that the O^{δ-} sites are used up after 15 minutes but some O₂⁻ species remain, implying that the former is more active than the latter. At the same time, the detection of the intermediates observed in the initial series of experiments occurs. This suggests that both O^{δ-} and O₂⁻ sites serve as active centres for propane oxidation, potentially exhibiting higher reactivity compared to the surface lattice O²⁻.

The fourth series: after cleansing the catalyst, the reaction flow consisting of 5000 ppm propane + 10% O₂/Ar was introduced into the cell at 300 °C. As shown in Fig. 10(d), after 5 minutes, the formation of the intermediates is obviously

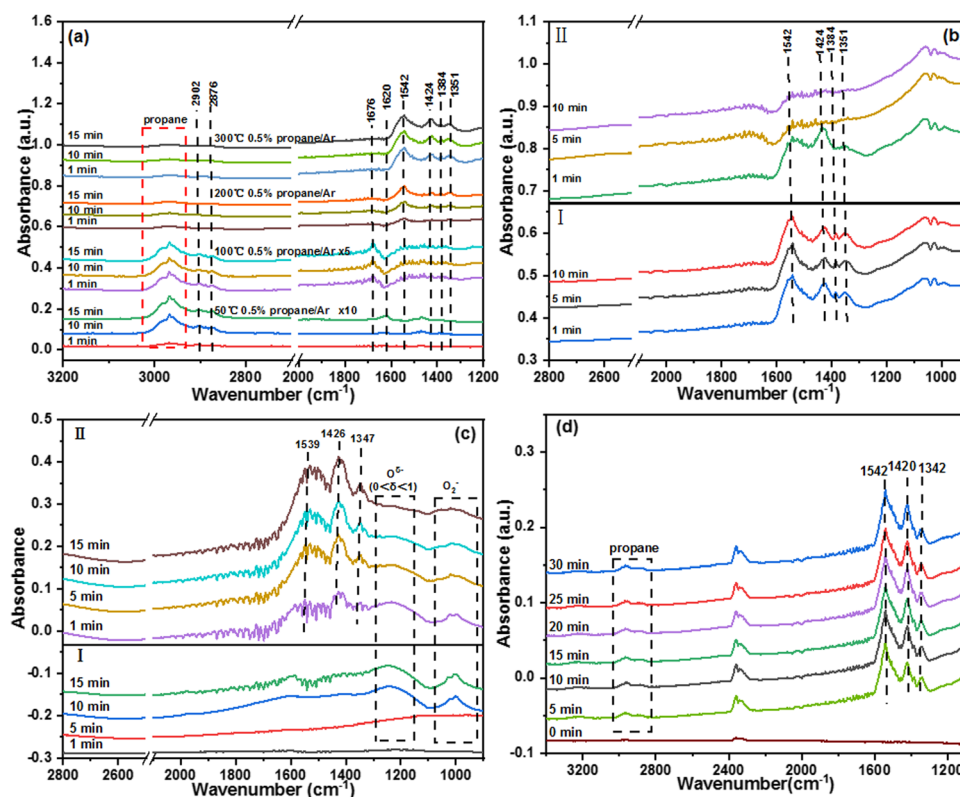


Fig. 10 *In situ* DRIFTS experiments on Cr–SnO₂, (a) exposed in propane/Ar flow under different situations, (b-I) exposed in propane/Ar flow at 300 °C, (b-II) exposed in 10% O₂/Ar flow at 300 °C, (c-I) exposed in a 10% O₂/Ar flow at 300 °C, (c-II) exposed in propane/Ar flow at 300 °C, (d) exposed in the reaction feed consisting of 5000 ppm propane/Ar + 10% O₂/Ar.

detected as well. In contrast, as shown in Fig. S5(a') and (b') (ESI†), the formation of the above intermediates on Mg-SnO₂ and Nb-SnO₂ is marginal under the same conditions, exhibiting that during the reaction process, propane molecules can only be mildly activated. Different from the second series of experiments, with the co-existence of propane and gaseous O₂, these intermediates are always detected. This strongly demonstrates that these surface species are the reaction intermediates for propane combustion, which can be formed, consumed, regenerated, and can reach equilibrium quickly under real reaction conditions with gas phase O₂.

Based on the *in situ* DRIFTS results, a possible pathway, propane → propene → acrylaldehyde → acrylic acid → acrylate → CO₃²⁻ → CO₂, is illustrated in Scheme 1.^{59,60,64} The adsorption of propane molecules on the Cr-SnO₂ surface is proposed to be converted into different intermediates, including propylene, acrolein, carboxylate, carbonate species and others. Both the adsorbed O^{δ-}/O₂⁻ species and lattice O²⁻ could be involved in the intermediate formation process, but the O^{δ-}/O₂⁻ anions can react more effectually with the intermediates to produce H₂O and CO₂.

3.10 Stability and tolerance tests for sulfur and/or water vapor

The reaction stability and sulphur and water resistance of Cr-SnO₂, the most active catalyst for propane combustion, have been tested at 320 °C. Fig. S6(a) (ESI†) shows that the propane conversion is steady in a dry feed during 50 hours' running. In addition, Fig. S6(b) (ESI†) exhibits that in 5% water vapor, the overall activity exhibits only a mild decrease but can be fully restored after removing water. Interestingly, adding only 10 ppm SO₂ in the flow has no negative impact on the activity, whereas adding 5% water vapor and 10 ppm SO₂ together can obviously degrade the activity, as evidenced by the decreased propane conversion. Furthermore, the deactivation cannot be restored completely after stopping the two poisoning agents.

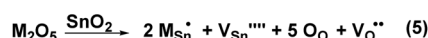
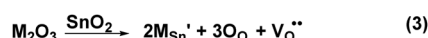
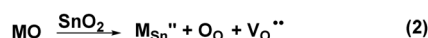
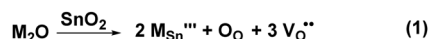
Therefore, Raman spectroscopy, FTIR spectroscopy and H₂-TPR techniques have been used to elucidate the reasons. In comparison with the fresh catalyst, Fig. S7(a) (ESI†) exhibits

that the characteristic Raman bands of SnO₂ phases disappeared on the partly deactivated catalyst, indicating that its surface composition has been changed. As exhibited in Fig. S7(b) (ESI†), a sulphate peak at 1000 cm⁻¹ is observed obviously *via* FTIR spectroscopy. Additionally, the H₂-TPR results in Fig. S7(c) (ESI†) have proven that the low temperature reducing peak belonging to the surface facile oxygen nearly disappears. This strongly shows that when water vapor and sulphur oxide are co-fed into the reaction flow, some stable surface sulphates are generated, which can damage the surface-active oxygen centres, thus being harmful to the catalytic reactivity.

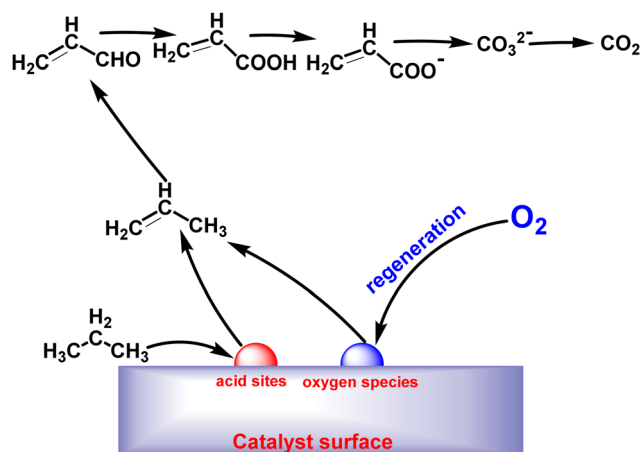
4. A short discussion

In this work, aiming to understand the effects of doping cations with different valence states on the structure–reactivity of solid solution catalysts, five metal ions (Li⁺, Mg²⁺, Cr³⁺, Zr⁴⁺ and Nb⁵⁺) with radii similar to Sn⁴⁺ (CN = 6) have been selected to modify SnO₂. As an endeavour to incorporate these cations into the lattice matrix of tetragonal SnO₂, a sol–gel method was employed to synthesize the catalysts. It was confirmed *via* XPS that the doping ions are present as Li⁺, Mg²⁺, Cr³⁺, Zr⁴⁺ and Nb⁵⁺ on/in the catalysts. XRD and Raman results have proven that Cr³⁺, Zr⁴⁺ and Nb⁵⁺ have successfully been doped into the lattice of SnO₂ to substitute part of the Sn⁴⁺ to generate solid solutions. As shown in the following equations (1–5), the incorporated ions (Cr³⁺, Zr⁴⁺ and Nb⁵⁺) typically differ from Sn⁴⁺ in terms of radius and/or valence state, hence resulting in charge imbalance, lattice distortion, and defects in the obtained solid solution. This ultimately enhances the diffusion rate of bulk phase oxygen and forms more oxygen vacancies/defects.

In turn, these defects/vacancies can promote the generation of a higher concentration of surface-active oxygen species (O^{δ-} and O₂⁻), which are favourable to CO oxidation and propane combustion.^{66–69} It is also discovered that for the three solid solution catalysts, if the doping cations (Cr³⁺ and Nb⁵⁺) have more varied valence states than Sn⁴⁺, there is a higher formation of surface defects than Zr⁴⁺ having the same valence state. Therefore, Nb-SnO₂ and Cr-SnO₂ possess a higher concentration of surface-active oxygen sites than Zr-SnO₂, displaying better oxidation activity.



However, although Li⁺ and Mg²⁺ have radii similar to Sn⁴⁺ at CN = 6, their valence state deviation is too big to form a solid solution structure, thus Li and Mg nitrates are present on the catalyst surfaces, which can cover and suppress the surface



Scheme 1 Proposed pathway for propane oxidation on Cr-SnO₂ catalysts.

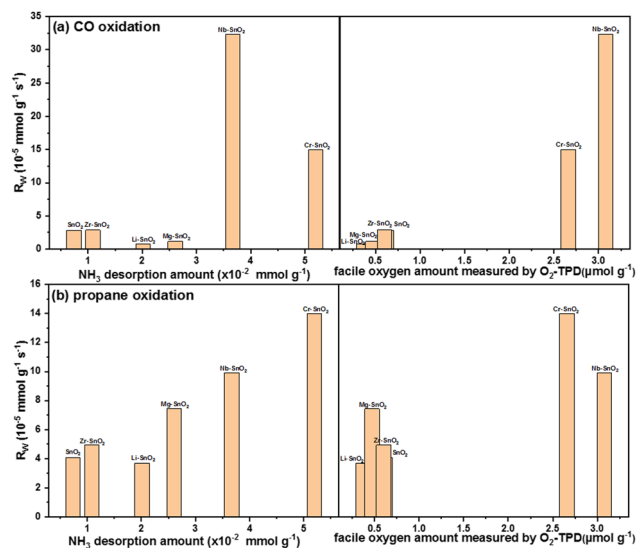


Fig. 11 (a) CO oxidation rates versus facile oxygen amount and NH_3 desorption amount; (b) propane oxidation rates versus facile oxygen amount and NH_3 desorption amount.

defects/vacancies of SnO_2 . As a result, Li-SnO_2 and Mg-SnO_2 exhibit lower oxidation activity than individual SnO_2 .

Most interestingly, though the concerted interaction between surface active oxygen and acidic sites is important for CO oxidation and propane oxidation, the two kinds of active sites have different weights in the two reactions. Fig. 11(a) displays that the activity of CO oxidation is almost positively correlated to the surface-active oxygen amount but not to the surface acid site amount. This indicates that the surface facile oxygen is more effective for CO oxidation than the surface acidity. However, Fig. 11(b) shows that the propane combustion activity is nearly proportional to the quantity of surface acid sites but not to the quantity of surface-active oxygen sites. This testifies that different from the case of CO oxidation, surface acid sites could play a more crucial role than the active oxygen species for propane combustion.

5. Conclusion

Targeting to unravel the valence state influence of doping cations on the structure-reactivity of solid solution catalysts, a series of metal ions (Li^+ , Mg^{2+} , Cr^{3+} , Zr^{4+} and Nb^{5+}) with radii similar to Sn^{4+} (CN = 6) but with varied valence states have been adopted to dope SnO_2 . The prepared catalysts have been characterized using multiple techniques and evaluated by CO and propane oxidation.

(1) XRD and Raman results have proven that Cr^{3+} , Zr^{4+} , and Nb^{5+} can be successfully incorporated into the matrix of SnO_2 to substitute part of the Sn^{4+} to form solid solutions, which results in more abundant surface defects. In contrast, due to too big deviation of the valence state from Sn^{4+} , Li^+ and Mg^{2+} cannot enter the lattice of SnO_2 but can mainly be present in the form of nitrates on the catalyst surfaces, thus suppressing the surface defects.

(2) The abundant surface defect on the solid solution catalysts promotes the generation of both active surface oxygen (O_2^- and $\text{O}^{\delta-}$) and acid sites, thus improving catalytic performance. It is concluded that the active oxygen sites play a more critical role than the acid sites for CO oxidation. In contrast, the surface acid sites play a more crucial role than the active oxygen sites for propane combustion.

(3) On the active Cr-SnO_2 for propane oxidation, reactive intermediates such as adsorbed propylene, acrylaldehyde, carboxylate *etc.* can be formed effectively. However, on those catalysts with low activity, the formation of these intermediates is negligible.

Author contributions

Haiming Yan: data curation, formal analysis, investigation, methodology, software, writing – original draft. Teng Liu: data curation, formal analysis, investigation, validation, writing – original draft. Yu Lv: data curation, investigation, validation. Xianglan Xu: data curation, investigation, methodology. Junwei Xu: data curation; investigation; methodology. Xiuzhong Fang: data curation, investigation, methodology. Xiang Wang: conceptualization, funding acquisition, investigation, methodology, project administration, resources, supervision, validation, writing – review & editing.

Conflicts of interest

There are no conflicts to declare.

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References

- 1 Z. Wang, Y. Liu, T. Yang, J. Deng, S. Xie and H. Dai, *Chin. J. Catal.*, 2017, **38**, 207–216.
- 2 C. He, J. Cheng, X. Zhang, M. Douthwaite, S. Patisson and Z. Hao, *Chem. Rev.*, 2019, **119**, 4471–4568.
- 3 J. Chen, X. Guo, L. Lang, X. Yin, A. Wang and Z. Rui, *Appl. Surf. Sci.*, 2023, **618**, 153275.
- 4 M. Ousmane, L. F. Liotta, G. Pantaleo, A. M. Venezia, G. Di Carlo, M. Aouine, L. Retailleau and A. Giroir-Fendler, *Catal. Today*, 2011, **176**, 7–13.
- 5 H. Huang, Y. Xu, Q. Feng and D. Y. C. Leung, *Catal. Sci. Technol.*, 2015, **5**, 2649–2669.
- 6 K. Yang, Y. Liu, J. Deng, X. Zhao, J. Yang, Z. Han, Z. Hou and H. Dai, *Appl. Catal., B*, 2019, **244**, 650–659.
- 7 Y. Guo, M. Wen, G. Li and T. An, *Appl. Catal., B*, 2021, **281**, 119447.

- 8 K. Zeng, Y. Wang, C. Huang, H. Liu, X. Liu, Z. Wang, J. Yu and C. Zhang, *Ind. Eng. Chem. Res.*, 2021, **60**, 6111–6120.
- 9 X. Ma, Y. Tang, Y. Liu, Y. Zhang, L. Jia, X. Liu, C. Du and B. Shan, *J. Catal.*, 2022, **409**, 59–69.
- 10 L. F. Liotta, *Appl. Catal., B*, 2010, **100**, 403–412.
- 11 Q. Zhang, S. Mo, J. Li, Y. Sun, M. Zhang, P. Chen, M. Fu, J. Wu, L. Chen and D. Ye, *Catal. Sci. Technol.*, 2019, **9**, 4538–4551.
- 12 T. F. Garetto, E. Rincón and C. R. Apesteguía, *Appl. Catal., B*, 2004, **48**, 167–174.
- 13 A. Hinz, M. Skoglundh, E. Fridell and A. Andersson, *J. Catal.*, 2001, **201**, 247–257.
- 14 Z. Hu, Z. Wang, Y. Guo, L. Wang, Y. Guo, J. Zhang and W. Zhan, *Environ. Sci. Technol.*, 2018, **52**, 9531–9541.
- 15 R. Peng, S. Li, X. Sun, Q. Ren, L. Chen, M. Fu, J. Wu and D. Ye, *Appl. Catal., B*, 2018, **220**, 462–470.
- 16 M. S. Kamal, S. A. Razzak and M. M. Hossain, *Atmos. Environ.*, 2016, **140**, 117–134.
- 17 L. Chen, J. Ding, J. Jia, R. Ran, C. Zhang and X. Song, *ACS Appl. Nano Mater.*, 2019, **2**, 4417–4426.
- 18 C. Feng, Q. Gao, G. Xiong, Y. Chen, Y. Pan, Z. Fei, Y. Li, Y. Lu, C. Liu and Y. Liu, *Appl. Catal., B*, 2022, **304**, 121005.
- 19 G. Li, N. Li, Y. Sun, Y. Qu, Z. Jiang, Z. Zhao, Z. Zhang, J. Cheng and Z. Hao, *Appl. Catal., B*, 2021, **282**, 119512.
- 20 C.-F. Liu, L.-C. He, X.-F. Wang, J. Chen, J.-Q. Lu and M.-F. Luo, *Mol. Catal.*, 2022, **524**, 112297.
- 21 S. Das and V. Jayaraman, *Prog. Mater. Sci.*, 2014, **66**, 112–255.
- 22 X. Kang, N. Deng, Z. Yan, Y. Pan, W. Sun and Y. Zhang, *Mater. Sci. Semicond. Process.*, 2022, **138**, 106246.
- 23 L. Villamagua, A. Stashans, P.-M. Lee, Y.-S. Liu, C.-Y. Liu and M. Carini, *Chem. Phys.*, 2015, **452**, 71–77.
- 24 Y. Guo, J. Liang, Y. Liu, Y. Liu, X. Xu, X. Fang, W. Zhong and X. Wang, *Ind. Eng. Chem. Res.*, 2019, **58**, 18569–18581.
- 25 Y. Guo, L. Zeng, X. Xu, Y. Liu, Y. Liu, X. Fu, Z. Gao, Z. Qian, J. Xu, X. Fang and X. Wang, *Appl. Catal., A*, 2020, **605**, 117755.
- 26 N. Kamiuchi, H. Muroyama, T. Matsui, R. Kikuchi and K. Eguchi, *Appl. Catal., A*, 2010, **379**, 148–154.
- 27 M. A. Mäki-Jaskari and T. T. Rantala, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2002, **65**, 1–8.
- 28 R. Sasikala, N. M. Gupta and S. K. Kulshreshtha, *Catal. Lett.*, 2001, **71**, 69–73.
- 29 J. Zhang, Y. Liu, Y. Sun, H. Peng, X. Xu, X. Fang, W. Liu, J. Liu and X. Wang, *Ind. Eng. Chem. Res.*, 2018, **57**, 10315–10326.
- 30 X. L. Xu, Y. Y. Tong, J. Y. Zhang, X. Z. Fang, J. W. Xu, F. Y. Liu, J. J. Liu, W. Zhong, O. E. Lebedeva and X. Wang, *Chin. J. Catal.*, 2020, **41**, 877–888.
- 31 X. L. Xu, F. Liu, X. Han, Y. Y. Wu, W. M. Liu, R. B. Zhang, N. Zhang and X. Wang, *Catal. Sci. Technol.*, 2016, **6**, 5280–5291.
- 32 X. Wang and Y. C. Xie, *Catal. Lett.*, 2001, **75**, 73–80.
- 33 X. Wang and Y. C. Xie, *Appl. Catal., B*, 2001, **35**, 85–94.
- 34 X. Yao, C. Tang, Z. Ji, Y. Dai, Y. Cao, F. Gao, L. Dong and Y. Chen, *Catal. Sci. Technol.*, 2013, **3**, 688–698.
- 35 Y. Zhang, Y. J. Zhou, J. P. Lin, G. L. Chen and P. K. Liaw, *Adv. Eng. Mater.*, 2008, **10**, 534–538.
- 36 A. R. Denton and N. W. Ashcroft, *Phys. Rev. A*, 1991, **43**, 3161–3164.
- 37 X. Feng, S. Zhang, R. Liu, J. Ma, X. Xu, J. Xu, X. Fang and X. Wang, *Phys. Chem. Chem. Phys.*, 2022, **24**, 3250–3258.
- 38 M. Balamurugan, R. Shanmugam, T.-W. Chen, S.-M. Chen, B.-S. Lou, X. Liu and C.-Y. Hong, *Food Chem.*, 2021, **361**, 130162.
- 39 Q. Sun, X. Xu, H. Peng, X. Fang, W. Liu, J. Ying, F. Yu and X. Wang, *Chin. J. Catal.*, 2016, **37**, 1293–1302.
- 40 Y.-C. Xie and Y.-Q. Tang, *Adv. Catal.*, Academic Press, 1990, vol. 37, pp. 1–43.
- 41 R. Zhang, X. Liu, H. Liang, X. Yang, J. Li, W. Ye, X. Wang and B. Liu, *RSC Adv.*, 2022, **12**, 26238–26244.
- 42 Z. Qu, M. Cheng, W. Huang and X. Bao, *J. Catal.*, 2005, **229**, 446–458.
- 43 Q. Zou, Y. Zhao, X. Jin, J. Fang, D. Li, K. Li, J. Lu and Y. Luo, *Appl. Surf. Sci.*, 2019, **494**, 1166–1176.
- 44 P. Venkataswamy, K. N. Rao, D. Jampaiah and B. M. Reddy, *Appl. Catal., B*, 2015, **162**, 122–132.
- 45 T. Li, G. Xiang, J. Zhuang and X. Wang, *Chem. Commun.*, 2011, **47**, 6060–6062.
- 46 Y. Gu, S. Shao, W. Sun, H. Xia, X. Gao, Q. Dai, W. Zhan and X. Wang, *J. Catal.*, 2019, **380**, 375–386.
- 47 X. Li, L. Zhang, Z. Yang, P. Wang, Y. Yan and J. Ran, *Sep. Purif. Technol.*, 2020, **235**, 116213.
- 48 A. Gurlo, *ChemPhysChem*, 2006, **7**, 2041–2052.
- 49 J. Liu, L. Zeng, X. Xu, J. Xu, X. Fang, Y. Bian and X. Wang, *ChemCatChem*, 2022, **14**, e2021019.
- 50 H. L. Wan, Z. S. Chao, W. Z. Weng, X. P. Zhou, J. X. Cai and K. R. Tsai, *Catal. Today*, 1996, **30**, 67–76.
- 51 B. Babu, I. N. Reddy, K. Yoo, D. Kim and J. Shim, *Mater. Lett.*, 2018, **221**, 211–215.
- 52 E. McCafferty and J. P. Wightman, *Surf. Interface Anal.*, 1998, **26**, 549–564.
- 53 V. J. Ferreira, P. Tavares, J. L. Figueiredo and J. L. Faria, *Catal. Commun.*, 2013, **42**, 50–53.
- 54 D. Farmanzadeh and A. Valipour, *Appl. Surf. Sci.*, 2018, **450**, 509–515.
- 55 J. Sun, Y. Lu, L. Zhang, C. Ge, C. Tang, H. Wan and L. Dong, *Ind. Eng. Chem. Res.*, 2017, **56**, 12101–12110.
- 56 H. Wang, Z. Qu, H. Xie, N. Maeda, L. Miao and Z. Wang, *J. Catal.*, 2016, **338**, 56–67.
- 57 J. Xu, B. Mojet, J. Vanommen and L. Lefferts, *J. Catal.*, 2005, **232**, 411–423.
- 58 J. Wang, Z. Yan, L. Liu, Y. Chen, Z. Zhang and X. Wang, *Appl. Surf. Sci.*, 2014, **313**, 660–669.
- 59 B. Wang, X. Wu, R. Ran, Z. Si and D. Weng, *J. Mol. Catal. A: Chem.*, 2012, **361–362**, 98–103.
- 60 B. Wang, X. Wu, R. Ran, Z. Si and D. Weng, *J. Mol. Catal. A: Chem.*, 2012, **356**, 100–105.
- 61 M. I. Z. M. A. Hasan and L. Pasupulety, *J. Phys. Chem. B*, 2002, **106**, 12747–12756.
- 62 D. D. Li, X.-Y. Leng, X.-F. Wang, H.-B. Yu, W.-Q. Zhang, J. Chen, J.-Q. Lu and M.-F. Luo, *J. Catal.*, 2022, **407**, 322–332.
- 63 W. L. S. Faria, C. A. C. Perez, D. V. César, L. C. Dieguez and M. Schmal, *Appl. Catal., B*, 2009, **92**, 217–224.

- 64 J. Wu, B. Chen, J. Yan, X. Zheng, X. Wang, W. Deng and Q. Dai, *Chem. Eng. J.*, 2022, **438**, 135501.
- 65 L. Cano-Casanova, B. Mei, G. Mul, M. A. Lillo-Rodenas and M. D. C. Roman-Martinez, *Nanomaterials*, 2020, **10**, 1314.
- 66 Z. Hu, B. Li, X. Sun and H. Metiu, *J. Phys. Chem. C*, 2011, **115**, 3065–3074.
- 67 M. D. Krcha, A. D. Mayernick and M. J. Janik, *J. Catal.*, 2012, **293**, 103–115.
- 68 A. P. Pushkar and J. J. Varghese, *J. Catal.*, 2022, **413**, 681–691.
- 69 X. Sun, B. Li and H. Metiu, *J. Phys. Chem. C*, 2013, **117**, 23597–23608.