



Cite this: *Phys. Chem. Chem. Phys.*,
2024, 26, 7580

An atomic layer deposition diffusion–reaction model for porous media with different particle geometries†

Niko Heikkinen, *^a Juha Lehtonen ^a and Riikka L. Puurunen ^b

This work presents a diffusion–reaction model for atomic layer deposition (ALD), which has been adapted to describe radial direction reactant transport and adsorption kinetics in a porous particle. Specifically, we present the effect of three particle geometries: spherical, cylindrical and a slab in the diffusion–reaction model. The reactant diffusion propagates as a unidimensional front inside the slab particle, whereas with cylinder and spherical particles, the reactant diffusion approaches the particle centre from two and three dimensions, respectively. Due to additional reactant propagation dimensions, cylindrical and spherical particles require less exposure for full particle penetration. In addition to the particle geometry effect, a sensitivity analysis was used to compare the impact of the particles' physical properties on the achieved penetration depth. The analysis evaluates properties, such as the combined porosity and tortuosity factor, mean pore diameter, specific surface area, pore volume, and particle radius. Furthermore, we address the impact of the reactant molar mass, growth-per-cycle (GPC), sticking probability, reactant exposure and deposition temperature on the simulated diffusion and surface coverage profiles. The diffusion–reaction model presented in this work is relevant for the design and optimization of ALD processes in porous media with different particle geometries.

Received 20th November 2023,
Accepted 6th February 2024

DOI: 10.1039/d3cp05639b

rsc.li/pccp

1. Introduction

Atomic layer deposition (ALD) can be used to prepare conformal coatings on porous media, such as aerogels,¹ molecular sieving membranes,² fuel cell components,³ carbon nanotubes⁴ and catalysts.⁵ The precise control of reactant deposition inside the porous media is achieved *via* self-limiting adsorption, regulated by available surface reaction sites, rather than the amount of introduced reactant.^{6–10} In catalyst preparation, ALD can be used to introduce active metals, promoters and other modifiers with tuneable coating thicknesses and penetration depths inside a porous particle.^{5,10–17} In order to achieve the desired reactant deposition result with different reactants and porous substrates, diffusion–reaction modelling can be used to design and optimize the ALD process. In the diffusion–reaction model, the particle geometry has an effect on the reactant transport inside the porous particle. Fig. 1 presents the particle geometries discussed in this work, where a spherical particle

geometry is relevant for heterogeneous catalyst applications, whereas the slab model could be used for porous membranes (*e.g.* fuel cell applications) and the cylinder geometry for porous high aspect ratio fibres and monoliths.

The literature presents various analytical and computational models for the ALD process in porous materials and well-defined high aspect ratio features. Gordon *et al.*¹⁸ present an analytical model to describe ALD growth inside high aspect ratio holes and trenches, where the reactant exposure required to achieve conformal coating increases quadratically with respect to the aspect ratio. This and other analytical models^{19–21} have been especially suitable for reactant propagation and surface reaction modelling in a diffusion limited growth setting. Computational models to address ALD conformality can be classified²² as ballistic transport–reaction,^{23,24} Monte Carlo^{21,25–28} and diffusion–reaction (sometimes “continuum”) models.^{1,4,29–31} Monte Carlo and diffusion–reaction models are applicable from continuum flow (Knudsen number, $Kn \ll 1$) to free molecular flow ($Kn \gg 1$) regimes, whereas ballistic transport–reaction models are used in the Knudsen diffusion regime (free molecular flow).^{24,30,31} Diffusion–reaction models are suitable for simplified particle geometries, while ballistic transport and Monte Carlo models are appropriate for complex structures.^{22,31}

The diffusion–reaction modelling of the ALD process in porous media is based on Fick's law of diffusion.^{32–34} In this work, we

^a VTT Technical Research Centre of Finland, P.O. Box 1000, FIN-02044 VTT, Espoo, Finland. E-mail: niko.heikkinen@vtt.fi, juha.lehtonen@vtt.fi

^b Department of Chemical and Metallurgical Engineering, Aalto University School of Chemical Engineering, Kemistintie 1, Espoo, Finland. E-mail: riikka.puurunen@aalto.fi

† Electronic supplementary information (ESI) available. See DOI: <https://doi.org/10.1039/d3cp05639b>



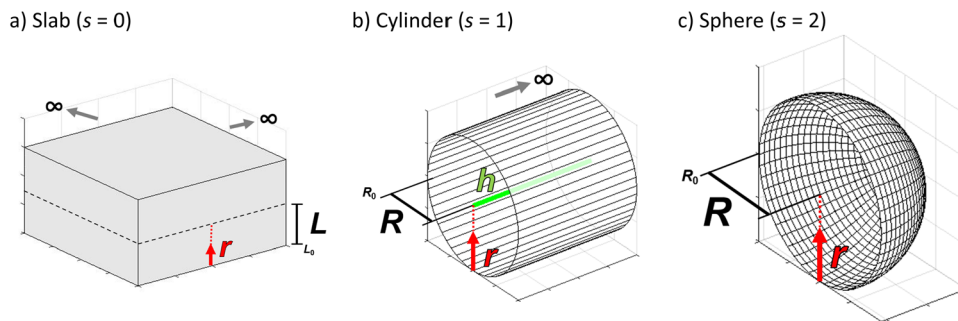


Fig. 1 Schematic illustrations of ideal particle geometries: (a) slab with a shape factor $s = 0$, (b) cylinder ($h = nR$, $n > 100$) with a shape factor $s = 1$ and (c) sphere with a shape factor $s = 2$.

assume homogeneous porosity and tortuosity throughout the particle, where reactant travel begins from the particle outer surface and the reactant diffuses through a tortuous pathway towards the centre of the particle. From the surface to the centre of the particle, the reactant travel is affected by the effective diffusion coefficient (D_e).^{35–37} Typically, the effective diffusion coefficient related to porous catalyst particles is determined in a Knudsen diffusion (D_{Kn}) regime as $D_e = D_{Kn}\epsilon\tau^{-1}$.^{35,36} With a Knudsen diffusion assumption, the effective diffusion coefficient depends on the pore dimensions, temperature and reactant molar mass,¹⁰ porosity (ϵ) and tortuosity (τ). The diffusion coefficient can also be used in a molecular diffusion regime or a combination of Knudsen diffusion and molecular diffusion.³⁵

Simplified particle geometries, which are especially relevant in diffusion–reaction modelling, are well known in the literature related to heterogeneous catalyst particles.^{35–37} The catalyst regeneration process is a typical example of gas diffusion and reaction in a spherical particle, where carbon formation (fouling) inside the catalyst particle is removed through oxidation.³⁵ Here, oxygen would be analogous to ALD reactant, diffusing into the catalyst pores and reacting with fouled carbon. Due to the spherical particle geometry, the oxygen demand decreases towards the particle centre. The diffusion–reaction regeneration process is commonly described with a shrinking core model (SCM), where the shell volume of a spherical particle decreases towards the particle centre.^{35,37}

In this work, a shape factor is used to determine partial differential equation for the respective particle geometry.^{36,38} Here, symbol r presents the radial coordinate along the characteristic dimension R . With cylindrical and spherical particles, R is the particle radius, whereas with slab particle geometry R is half of the slab thickness. The radial coordinate r is defined to be zero at the outer surface of a particle, and R at the centre for a sphere/cylinder or half thickness for a slab particle. With the slab particle, the ALD reactant is propagating from under- and upside directions towards the centre of the slab. With the cylinder particle, the reactant propagates in a radial direction, whereas the axial propagation direction is neglected. With the spherical particle, the reactant is propagating from all directions towards the centre. In the literature, the shape factor is determined as the external surface area divided by the particle volume, which is then multiplied by the particle radius R .^{35,37}

The shape factor is a common simplification to describe the effect of the particle geometry on a gaseous and liquid flow through a packed catalyst bed (for example, Thoenes–Kramers mass transfer correlation).³⁵ Furthermore, relevant to our work, the shape factor can be also used to describe particle geometry effect on the intraparticle gas diffusion.³⁷ Most relevant previous work on the particle geometry effect on the intraparticle diffusion, specifically in the ALD process, has been presented by Lee *et al.*³⁹ They presented a model to address particle geometry effect on the minimum exposure time for the complete ZnO coverage of planar, cylindrical and spherical porous alumina monoliths. Additionally, Elam *et al.*⁶ and Detavernier *et al.*⁴⁰ have also shown that the spherical particle geometry has a significant effect on the decreasing reactant consumption towards the particle centre.

Our work adapts slab, cylinder and spherical shape factors in an ALD diffusion–reaction model. With partial differential equations for different particle geometries, we demonstrate the effect of the shape factor on the diffusion and surface coverage profiles. We discuss the radial reactant propagation for each particle geometry and evaluate the effect of ALD process parameters and porous material properties on the achieved coating penetration depth. We implement parameters for the growth-per-cycle (GPC, from areal number density of the metal adsorbed), sticking probability, particle porosity and tortuosity, pore diameter, specific surface area, pore volume and particle radius into the coupled partial differential equations. The modelling principles in this work might be relevant also to thin film preparation methods beyond ALD, for example, sequential infiltration synthesis⁴¹ and vapor-phase infiltration.⁴²

2. Methods

2.1 Shape factors and diffusion–reaction differential equations

The shape factor (see Fig. 1) is relevant for modelling of diffusion in a porous material, as the modelled reactant consumption profiles vary according to the used shape factors. Eqn (1) presents the diffusion–reaction equation for the volumetric reactant number density (sometimes “particle number density”) with respect to the deposition time for the different



shape factors ($s = 0, 1, 2$).^{34,37}

$$\frac{\partial n_{\text{P}}}{\partial t} = \frac{1}{r^s} \frac{\partial}{\partial r} \left[D_{\text{c}} r^s \frac{\partial n_{\text{P}}}{\partial r} \right] - \frac{1}{4} v_{\text{th}} \beta_0 \cdot n_{\text{P}}(t, r) \cdot [1 - \theta(t, r)] \quad (1)$$

Here, n_{P} (m^{-3}) is the volumetric reactant number density and t (s) is the reactant exposure time. In this work we use the common concept that the reactant exposure (Pa s) is the partial pressure multiplied by the reactant exposure time.^{7,31,43} The symbol r presents the radial coordinate along the characteristic dimension R . In eqn (1), the adsorption loss term includes $\bar{s}_4 = A_0/V_{\text{p}}$ (m^{-1}) the ratio of specific surface area ($\text{m}^2 \text{g}^{-1}$) and pore volume ($\text{m}^3 \text{g}^{-1}$), v_{th} ($\text{m} \text{s}^{-1}$) the thermal velocity of the reactant in the gas phase, β_0 the reactant sticking probability and θ the surface coverage (fraction of occupied adsorption sites) at time t and position r inside the catalyst particle.

In the eqn (1), the symbol s is the shape factor with a relationship $s = a - 1$, where factor a has values of one, two and three for a slab, cylinder and sphere, respectively.³⁵⁻³⁷ The factor a is $(A_{\text{Par}}/V_{\text{Par}})R$,^{35,36} where the particle radius R (m) is multiplied by the ratio of the particle outer surface area (A_{Par} , m^2) and the catalyst particle volume (V_{Par} , m^3). Factor a related to the slab particle geometry is $(A_{\text{Par}}/V_{\text{Par}})L$, where L is the half thickness of the slab, corresponding to $L = R/2$. Table S1, ESI† presents some example calculations to determine the shape factor s for relevant particle geometries. (Note: one should not confuse A_{Par} and V_{Par} with the specific surface area (A_0 , $\text{m}^2 \text{g}^{-1}$) and pore volume (V_{p} , $\text{m}^3 \text{g}^{-1}$) of the porous media.) The ratio $A_{\text{Par}}/V_{\text{Par}}$ represents the surface area per volume of a non-porous particle. The adsorption loss term incorporates the self-limiting nature of the ALD process in the diffusion–reaction model.

The effective diffusion coefficient (D_{e} , $\text{m}^2 \text{s}^{-1}$), is derived from the gas phase diffusion coefficient in porous materials, consisting of both molecular diffusion and Knudsen diffusion factors. In eqn (2), according to the Bosanquet relation, the gas phase diffusion coefficient (D_{i}) is a sum of the molecular diffusion coefficient (D_{mi}) and Knudsen diffusion coefficient (D_{Kn}).²⁵

$$\frac{1}{D_{\text{i}}} = \frac{1}{D_{\text{mi}}} + \frac{1}{D_{\text{Kn}}} \quad (2)$$

In this work, the transportation of reactant molecules is assumed to be dominated by Knudsen diffusion (the reactant mean free path is much larger compared to the limiting pore dimensions). Then, the term for D_{mi} is neglected:

$$\frac{1}{D_{\text{i}}} = \frac{1}{D_{\text{Kn}}} \rightarrow D_{\text{i}} = D_{\text{Kn}} \quad (3)$$

In porous materials, the gas diffusion is affected by the characteristic pore structure.⁴⁴ To account for the intraparticle tortuosity (τ) and porosity (ε), the gas phase diffusion coefficient D_{i} has to be corrected with a factor $\varepsilon \tau^{-1}$.^{35-37,45} The porosity and tortuosity are used to estimate the fraction of the void space available for gas diffusion and the complexity of the

diffusion pathway, respectively.³⁶ The effective diffusion coefficient (D_{e}) is then:

$$D_{\text{e}} = \frac{\varepsilon}{\tau} D_{\text{Kn}} \quad (4)$$

After substituting the Knudsen diffusion coefficient (D_{Kn})³⁵⁻³⁷ according to the kinetic theory of gases into eqn (4), the effective diffusion coefficient (D_{e}) becomes:

$$D_{\text{e}} = \frac{\varepsilon}{\tau} D_{\text{Kn}} = \frac{\varepsilon d_{\text{pore}}}{\tau} \sqrt{\frac{8k_{\text{B}}T}{\pi m_{\text{r}}}} \quad (5)$$

where d_{pore} is the catalyst pore diameter (m), k_{B} is the Boltzmann constant ($\text{m}^2 \text{kg} \text{s}^{-2} \text{K}^{-1}$), T is the ALD process temperature (K) and m_{r} is the mass for a single reactant molecule (g).

After introducing the corresponding shape factor values to the eqn (1), the final partial differential equations for a slab (eqn (6)), cylinder (eqn (7)) and sphere (eqn (8)) can be written in the form:

(Slab, $s = 0$)

$$\frac{\partial n_{\text{P}}(t, r)}{\partial t} = D_{\text{e}} \frac{\partial^2 n_{\text{P}}(t, r)}{\partial r^2} - \frac{1}{4} v_{\text{th}} \beta_0 \cdot n_{\text{P}}(t, r) \cdot [1 - \theta(t, r)] \quad (6)$$

(Cylinder, $s = 1$)

$$\frac{\partial n_{\text{P}}(t, r)}{\partial t} = D_{\text{e}} \frac{\partial^2 n_{\text{P}}(t, r)}{\partial r^2} + D_{\text{e}} \frac{1}{R} \frac{\partial n_{\text{P}}(t, r)}{\partial r} - \frac{1}{4} v_{\text{th}} \beta_0 \cdot n_{\text{P}}(t, r) \cdot [1 - \theta(t, r)] \quad (7)$$

(Spherical, $s = 2$)

$$\frac{\partial n_{\text{P}}(t, r)}{\partial t} = D_{\text{e}} \frac{\partial^2 n_{\text{P}}(t, r)}{\partial r^2} + D_{\text{e}} \frac{2}{R} \frac{\partial n_{\text{P}}(t, r)}{\partial r} - \frac{1}{4} v_{\text{th}} \beta_0 \cdot n_{\text{P}}(t, r) \cdot [1 - \theta(t, r)] \quad (8)$$

In order to solve the partial differential equations for $\partial n_{\text{P}}/\partial t$, the coupled equation for surface coverage $\partial \theta/\partial t$ is determined with eqn (9), where the reactant adsorption kinetics are described by irreversible single-site Langmuir adsorption.^{22,32}

$$\frac{\partial \theta(t, r)}{\partial t} = -s_0 \frac{1}{4} v_{\text{th}} \beta_0 \cdot n_{\text{P}}(t, r) \cdot [1 - \theta(t, r)] \quad (9)$$

s_0 is the surface area of adsorption sites (m^2) calculated from the reciprocal of σ_{P} ($s_0 = 1/\sigma_{\text{P}}$). Here, σ_{P} is the areal number density (m^{-2}) of metal atoms deposited on each exposure cycle at saturation. The σ_{P} directly related to the growth-per-cycle (GPC) in ALD⁷ and s_0 is then inversely related to the GPC.^{22,46} The diffusion–reaction model in this work uses the volumetric reactant number density (n_{P}) for the initial reactant concentration. The ideal gas law is used to determine the volumetric reactant number density from the reactant partial pressure and deposition temperature. The exposure has a unit Pa s, derived from the reactant partial pressure (Pascal) and the reactant exposure time (seconds).

The model initial condition for θ is presented in eqn (10). At time $t = 0$, θ is zero for all values of the radial distance (r) within the range from 0 to R .

$$\theta(t = 0, r) = 0, \forall r \in [0, R \mu\text{m}] \quad (10)$$



Eqn (11) gives the initial condition for n_p . At time $t = 0$, n_p is zero for all values of the radial distance (r) including all real numbers greater than 0 and less than or equal to R .

$$n_p(t = 0, r) = 0, \forall r \in [0, R \text{ }\mu\text{m}] \quad (11)$$

As the diffusion–reaction model contains a second-order differential equation, two boundary conditions are required. The Dirichlet boundary condition^{35,47} in eqn (12) presents that any given time t , the volumetric reactant number density at the particle surface ($r = 0$) is equal to a constant value n_{p0} (the initial reactant volumetric reactant number density).

$$n_p(t, r = 0) = n_{p0}, \forall t \geq 0 \quad (12)$$

The Neumann boundary condition^{35,47} in eqn (13) shows that at the particle centre ($r = R$), the rate of change of the volumetric reactant number density with respect to r is zero. This boundary condition sets a limit to the particle centre, wherefrom particles are unable to pass through.

$$\left. \frac{\partial n_p(t, r)}{\partial r} \right|_{r=R \text{ }\mu\text{m}} = 0, \forall t \geq 0 \quad (13)$$

Partial differential equations (PDEs) were solved in Matlab (MATLAB Version: 9.14.0 (R2023a)) by discretizing the first and second derivatives of the particle radius applying finite differences. For the first derivative, dss004 code was applied using five-point fourth order approximation and for the second derivative, dss044 code using six-point fourth order approximation was used.⁴⁸ The spatial domain of the radius was discretized typically using 300 points. The formed system of ODEs was solved using Matlab solver ode15s being applicable for stiff ODE systems.

2.2 ALD reactant and catalyst structural parameters

The above presented coupled differential equations (see eqn (6)–(9)) can be used to determine the ALD reactant consumption and surface coverage profiles. The resulting profiles will be influenced by parameters related to the reactant molecule, structural properties of the porous media and the ALD process. Table 1 presents the baseline parameters for the porous media and Table 2 shows the baseline parameters for the ALD process used in the simulations in this work. These baseline parameters were used to compare the effect of particle shape factors on the resulting reactant consumption and surface coverage profiles. While the values do not represent any specific ALD reactant or porous media, they have been inspired by a typical ALD process, such as the trimethylaluminium–water process⁴⁹

Table 1 Baseline parameters for porous media

Parameter	Value
Porosity, ε	0.5
Tortuosity, τ	1
Mean pore diameter, d_{pore}	2×10^{-8} m
Specific surface area, A_0	$200 \text{ m}^2 \text{ g}^{-1}$
Pore volume, V_p	$6 \times 10^{-7} \text{ m}^3 \text{ g}^{-1}$
Particle radius, R	3×10^{-4} m

Table 2 Baseline parameters for ALD process and reactant

Parameter	Value
Reactant molar mass, m_R	100 g mol^{-1}
Initial sticking probability, β_0	2×10^{-2}
Areal number density of metal atoms deposited on each exposure cycle at saturation, σ_p	$4 \times 10^{18} \text{ m}^{-2}$
Reactant volumetric number density outside the porous material, n_{p0} ($t, r = 0$)	$1.71 \times 10^{22} \text{ m}^{-3 a}$
Temperature, T	423 K
Reactant exposure time, t	100, 500 and 1000 s^b

^a Estimated from ALD reactant vapour pressure of 100 Pa at 423 K.
^b Baseline reactant exposure duration marked with bold text.

and by mesoporous catalyst supports,⁵⁰ such as silica or alumina.²⁹

3. Results and discussion

3.1 The effect of the shape factor on the volumetric reactant number density and surface coverage profiles

The selection of the shape factor has a distinctive effect on the diffusion–reaction model simulation results. Fig. 2 presents simulated profiles for the volumetric reactant number density and surface coverage as a function of radial coordinate r (r is zero at the particle surface and increases until the centre of the particle, where $r = R$). Here, the coating penetration depth varies according to the particle geometry. The penetration depth ($r_{\theta=1/2}$) in Fig. 2(a) and (b) was 177, 133 and 114 μm for the spherical ($s = 2$), cylindrical ($s = 1$) and slab ($s = 0$) geometries, respectively. In the panel (a), the reactant number density profile for the slab shape factor shows known decreasing linear trend as a function of the penetration depth,^{18,39} whereas profiles for the cylinder and sphere are curved. In general, for the same exposure, the ALD coating penetrates deeper in the order of sphere > cylinder > slab. In this work, the shape factor $s = 1$ considers long cylinders, where the dimension h is large with respect to R (see Fig. 1). A cylinder with $h = nR$, $n \geq 2$, the shape factor $s = a - 1 = \frac{2(n+1)}{n} - 1$, where the effect of the cylinder bottom and top decrease when n increases.³⁶ With increasing values of n , the shape factor (S) approaches the value 1. If n is small, the shape factor approaches a value of 2 and the partial differential equation $\partial n_p / \partial t$ (eqn (7)) becomes similar to spherical particle geometry (eqn (8)). For values $n < 2$, the shape factor for cylindrical geometry is not defined.³⁶

Fig. 2(a) presents the particle geometry effect on the gas-phase reactant number density as a function of the coating penetration depth. The slab geometry demonstrates a linear behaviour as the reactant diffuses through a “plate” with an infinite depth and width, with only one diffusion propagation dimension.^{35,36} The cylinder profile is curved because of two diffusion propagation dimensions and one indefinite dimension (see Fig. 1). The spherical geometry has no infinite dimensions and the reactant diffusion front approaches the



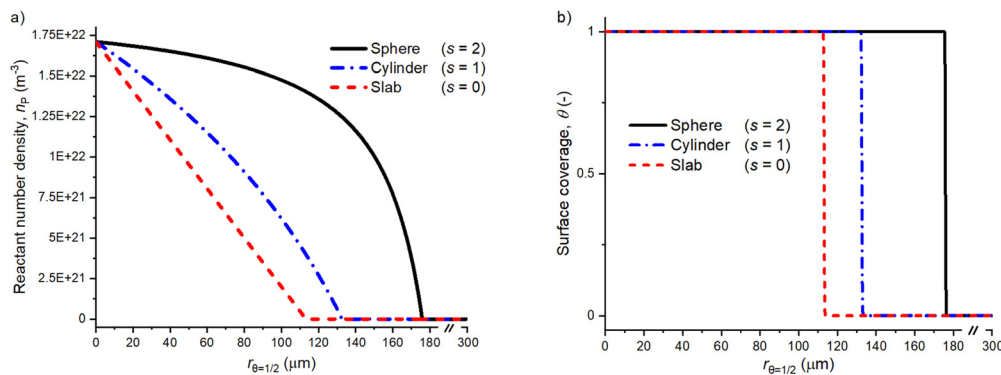


Fig. 2 The effect of the shape factor (s) as a function of the radial coordinate r for (a) the volumetric reactant number density and (b) the surface coverage. A list of the used parameters is presented in Tables 1 and 2.

particle centre from all directions. Then, the deposition area per shell volume decreases towards the centre of the porous particle.³⁵

Although the shape factor is not included in the equation for the surface coverage (eqn (9)), the cylinder and spherical particle penetration depth increases in the surface coverage profile compared to the slab in Fig. 2(b). This is due to $\partial n_p(t,r)/\partial t$ and $\partial \theta(t,r)/\partial t$ equation coupling. Despite the increasing penetration depth for the cylinder and spherical shape factors compared to the slab, all three surface coverage profiles have a similar overall shape and slope at the adsorption front.

3.2 Shape factor and exposure needed for full particle penetration

The shape factor has an impact on the achieved penetration depth ($r_{\theta=1/2}$) for a given exposure. The value for the penetration depth ($r_{\theta=1/2}$) is determined as the position where the surface coverage $\theta = 0.5$.^{32,51} Fig. 3 presents the penetration depths

obtained from simulations in which all the parameters remain the same, except that the porous media particle radius R is varied. For example, the penetration depth values from Fig. 1 (177 μm and 114 μm) can be seen in Fig. 3 in red at $R = 200$ μm (the red sphere marker indicates spherical particle geometry and a red rectangle shows the slab geometry).

Fig. 3 presents the penetration depth results for each shape factor as a function of the particle radius (R). The distinct line starting from the origin is the penetration depth, where the reactant has reached the centre of the particle with a radius R . For example, for a particle with a radius of 50 μm, the penetration depth is 50 μm for all the presented exposures and shape factors. With an increased exposure, larger particles can be fully penetrated by the ALD reactant. In Fig. 3, full penetration is achieved with a 1×10^5 Pa s exposure and a spherical shape factor when $R \leq 275$ μm. A reduced particle size can be penetrated with the same exposure with $R \leq 225$ μm and ≤ 161 μm for the cylinder and slab particle geometries, respectively. According to Fig. 3, when the particle radius increases, all shape factors result in increasingly similar values for the penetration depth. This behaviour is expected, as the increasing radius of a particle reduces the factor $1/R$ and $2/R$ in eqn (7) and (8). Increasing the particle radius R will diminish the effect from the first order partial derivative, resulting in an asymptotically closing equation with respect to the slab geometry (see eqn (6)).

In the example in Fig. 3, the shape factor has a significant effect on the modelled penetration depth when particle radius is < 600 μm. However, the model results are not bound to a specific particle size. Instead, the important particle size range depends on the used exposure. Fig. 4 presents the penetration depth difference between slab and spherical geometries. In the figure, the horizontal axis is the normalized penetration depth for sphere, $\zeta_{\theta=1/2, \text{sphere}} = \frac{r_{\theta=1/2, \text{sphere}}}{R}$, where the value 1 indicates a full coating (e.g., when particle radius is 200 μm and the reactant has reached the centre of the particle at 200 μm, $R = 200$ μm and $r_{\theta=1/2, \text{sphere}} = 200$ μm). The vertical axis in the figure is the percentage difference between the penetration depths in the slab and spherical geometry. These values can be found in Fig. 3, where for example, a 1×10^5 Pa s exposure resulted in $r_{\theta=1/2, \text{sphere}}$

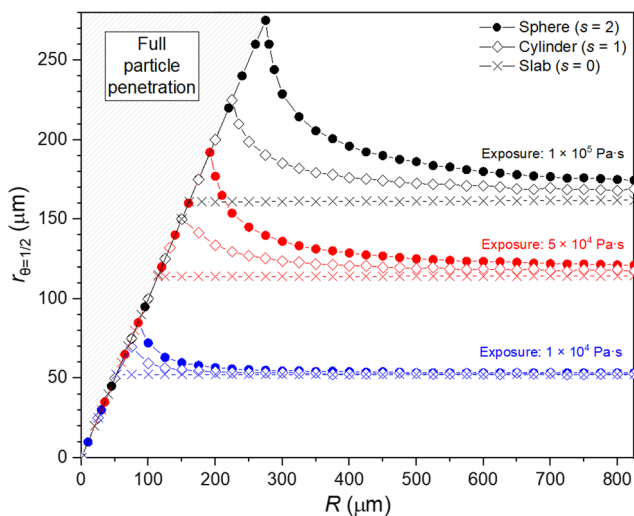


Fig. 3 The effect of varying particle size R on the achieved $r_{\theta=1/2}$ (penetration depth) with three different exposures of reactant (1×10^4 , 5×10^4 and 1×10^5 Pa s). Three curves for each exposure present the model results for different shape factors: sphere ($s = 2$), cylinder ($s = 1$) and slab ($s = 0$). The simulation parameters are presented in Tables 1 and 2.



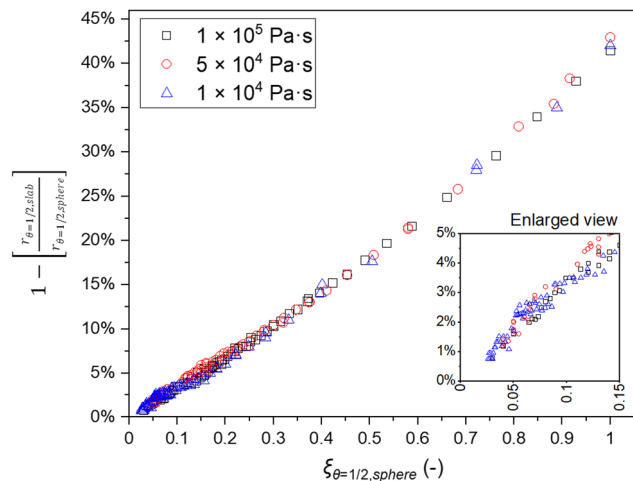


Fig. 4 The percentage difference between the penetration depths of the slab and spherical shape factors as a function of $\xi_{\theta=1/2}$. The vertical axis is the percentage difference in the penetration depth between the slab and spherical shape factors. The horizontal axis presents the normalized value ξ of penetration depth with the spherical shape factor per particle radius. The small inset is an enlarged view of the same plot. The figure shows the results for 1×10^4 , 5×10^4 and 1×10^5 Pa s exposures at 423 K and the particle radius (R) varying from 50 to 3000 μm in 25 μm step intervals. All other parameters are presented in Tables 1 and 2.

of 229 μm and $r_{\theta=1/2, \text{slab}}$ of 161 μm . In Fig. 4, these values can be found from the vertical axis percentage difference of $30\% = 1 -$

$$\frac{161 \mu\text{m}}{229 \mu\text{m}} \text{ and horizontal axis value } \xi_{\theta=1/2, \text{sphere}} \text{ of } 0.76 = \frac{229 \mu\text{m}}{300 \mu\text{m}}.$$

Fig. 4 illustrates the difference in the achieved penetration depth between the slab and spherical shape factors. When $\xi_{\theta=1/2, \text{sphere}} \leq 0.15$, the percentage difference in the penetration depth estimate between the slab and spherical shape factors is $< 5\%$. In order to achieve a $< 1\%$ difference, $\xi_{\theta=1/2, \text{sphere}}$ should be ≤ 0.05 . Increasing the penetration depth with respect to the particle radius, the selection of the shape factor becomes increasingly relevant. At full infiltration, the penetration depth reaches $\sim 40\text{--}45\%$ difference. This finding corresponds to the work by Lee *et al.*,³⁹ Elam *et al.*⁶ and Detavernier *et al.*,⁴⁰ in which the spherical particle geometry has a significant effect on the decreasing reactant consumption towards the particle centre.

3.3 Sensitivity analysis: baseline parameters with one varying parameter

A sensitivity analysis was conducted to address the effect of varying a single parameter at a time on the volumetric reactant number density and surface coverage profiles. Fig. 5 presents the volumetric reactant number density profiles for the spherical particle geometry (left) and slab particle geometry profiles (right). In all figures, the solid black line illustrates the baseline parameters and other profiles are for varied parameter values.

Fig. 5 panels (a) and (b) show the effect of tortuosity (τ) and porosity (ϵ) as a single factor τ/ϵ for the sphere and slab respectively. The factor τ/ϵ is a reciprocal value of ϵ/τ which is

present in the effective diffusion coefficient (see eqn (5)). Factor values from 1 to 5 are typical for heterogeneous catalyst support materials, where increasing tortuosity increases the diffusion, *i.e.* the precursor travel distance inside the porous particle. Similarly, reducing the porosity increases the diffusion resistance due to the retained reactant passage through the porous media. Interestingly, with a restricted diffusion and increasing τ/ϵ , the achieved $r_{\theta=1/2}$ is increasingly similar between slab and spherical particles. With a decreasing τ/ϵ the difference between the particle geometries becomes apparent, especially with the shrinking core effect from the spherical particle geometry.³⁵

Panels (c) and (d) in Fig. 5 present the effect of the reactant molar mass on the penetration depth ($r_{\theta=1/2}$). The penetration depth decreases with an increasing reactant molar mass due to the decreasing reactant gas-phase mean thermal velocity v_{th} (m s^{-1}). This observation corresponds to the findings reported by Yim *et al.*⁴³

Panels (e) and (f) in Fig. 5 illustrate the effect of the areal number density (σ_{p} , m^{-2}) of deposited metal atoms upon saturation (directly related to the GPC) on the penetration depth. Here, smaller values increase penetration depth due to decreased amount of reactant adsorption sites. The presented trend is similar to Yim *et al.*,⁴³ where an increasing σ (nm^{-2}) strongly reduces the achieved $r_{\theta=1/2}$. With $\sigma = 1 \text{ nm}^{-2}$, the spheres have been fully coated.

Panels (g) and (h) in Fig. 5 present the effect of varying $\bar{s} = A_{\text{O}}/V_{\text{p}}$ (m^{-1}), a factor found in eqn (6)–(8). Here, increasing the ratio \bar{s} increases the surface area compared to the pore volume. Thus, the increased surface area per given volume (increasing \bar{s}) results in an increased number of surface adsorption sites per volume and a reduced penetration depth.

Panels (i) and (j) in Fig. 5 present the volumetric reactant number density profiles that appear the same for all varied values of β_0 . Similar findings have been reported by Keuter *et al.*³⁴ and Arts *et al.*,⁵¹ where varying the β_0 value has no significant effect on the average coating penetration depth. However, the enlarged subplot reveals minute differences in the profiles with varying β_0 . When the reactant number density approaches zero (reactant depletion), the decreasing sticking probability (β_0) reduces the rate of reactant adsorption on the surface, enabling the reactant to travel deeper into the particle. On the other hand, different findings have been reported by Yim *et al.*,⁴³ where they observed a slight increase in the $r_{\theta=1/2}$, as β_0 increased. Yim *et al.*⁴³ assumed that this would be a result of their analytical model approximation for the reactant partial pressure, resulting in a penetration depth ($r_{\theta=1/2}$) increase, when the β_0 value increases.

Lastly, panels (k) and (l) in Fig. 5 present profiles with varied pore diameters. In the presented diffusion–reaction model, the pore diameter is only present in the Knudsen diffusion coefficient (see eqn (5)). A reduced pore diameter restricts diffusion and results in a decreased penetration depth. For both the slab and spherical particle, the $r_{\theta=1/2}$ increases linearly with the pore diameter. However, due to the spherical particle geometry, the



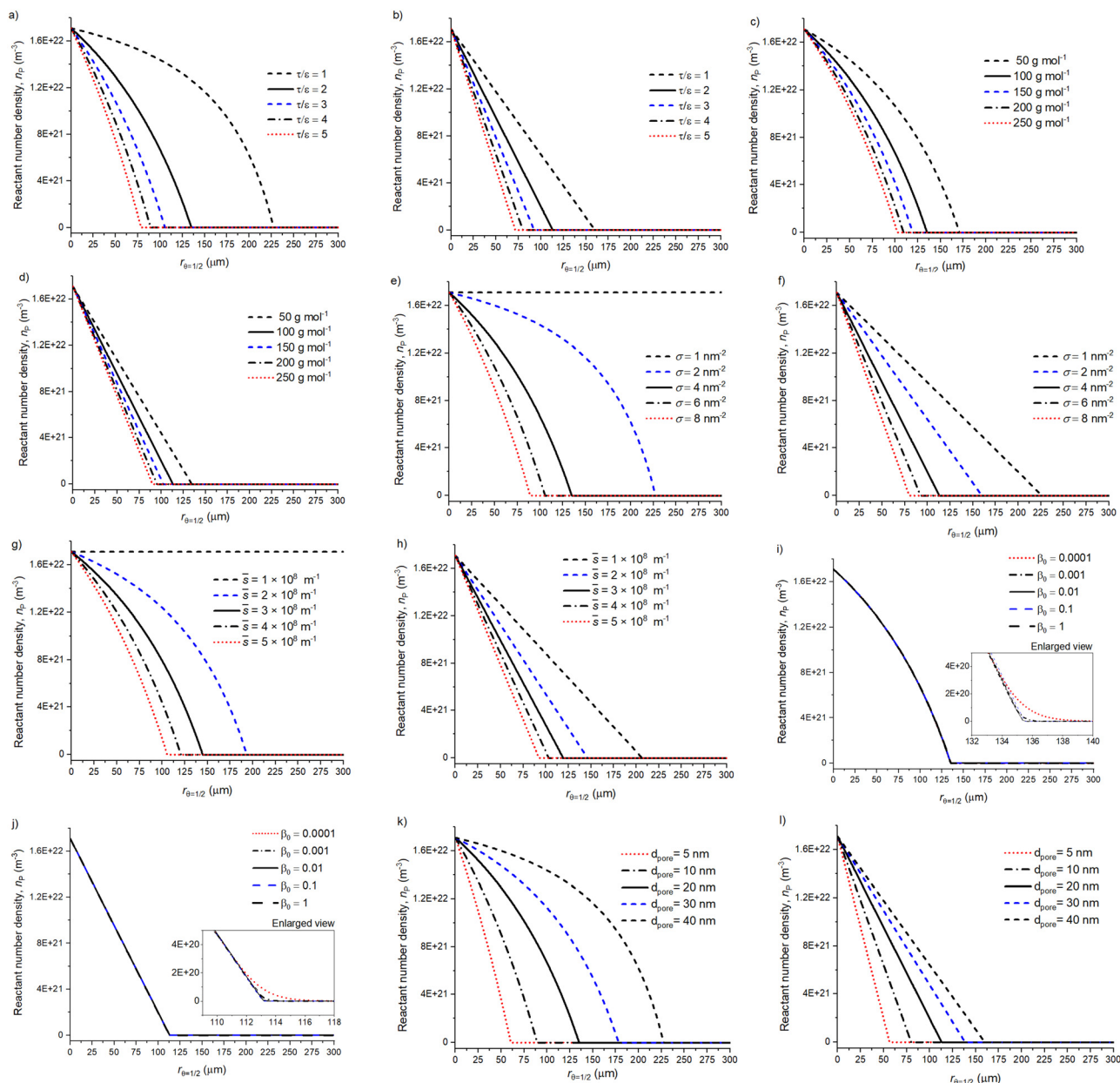


Fig. 5 Parameter effect on the volumetric reactant number density profile (n_p) with a spherical shape factor and slab shape factor. Panel (a) and (b) show the effect of tortuosity (τ) per porosity (ϵ), (c) and (d) reactant molar mass variation (m_r), (e) and (f) show the reactant areal number density (σ , atoms per nm^2), (g) and (h) varying $\bar{S} = A_G/V_p$ (m^{-1}), (i) and (j) sticking probability (β_0), (k) and (l) pore diameter (d_{pore} , m). The parameter effect is addressed by changing one parameter at a time and all other values are baseline parameters with a 500 s reactant exposure duration (see Table 1 and 2). The related surface coverage profiles are shown in Fig. 6.

$r_{\theta=1/2}$ increases faster. For example, increasing the d_{pore} from 10 to 40 nm will double the $r_{\theta=1/2}$ for the slab. The same increase with spherical particle increases the $r_{\theta=1/2}$ 2.5 times.

Fig. 6 presents the surface coverage profiles connected to the volumetric reactant number density profiles in Fig. 5. For example, panels (a) and (b) illustrate the effect of the τ/ϵ on the surface coverage profile. Panels (i) and (j), show the β_0 surface coverage profiles revealing a widening distribution as the β_0 decreases. Interestingly, the spherical and slab particle geometries seem to have similar surface coverage profiles with

respect to a varied β_0 , and only the penetration depth is different. Similar surface coverage profiles for the slab and spherical particle geometries with a varying β_0 are due to the model using the same ALD process, reactant and particle properties for both the spherical and slab particle geometries. Then, the only difference is the reactant propagation inside the particles. With the spherical particle, the reactant propagation depends on the location (radial coordinate) inside the particle, whereas the slab particle reactant propagation is independent of the location.



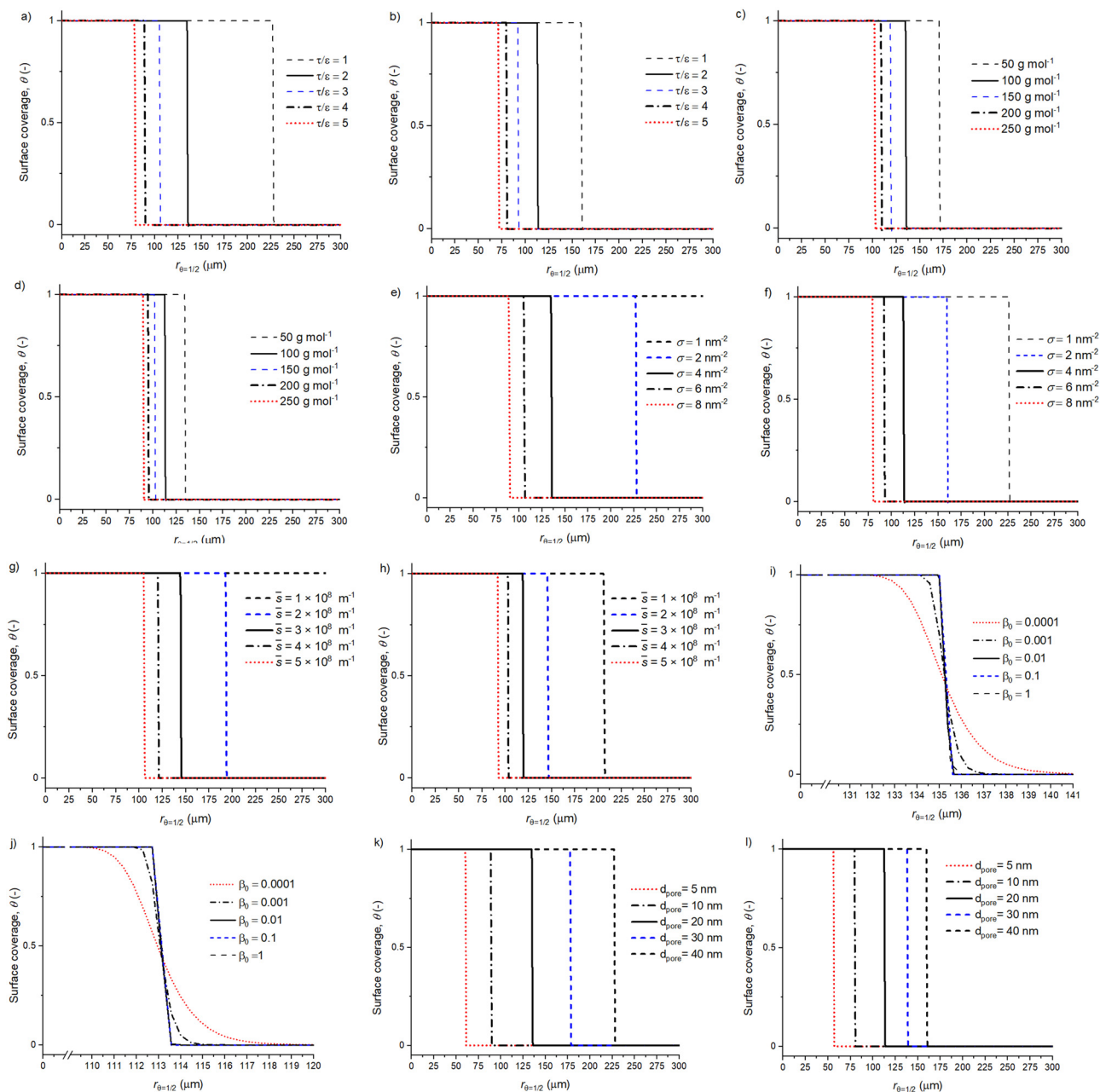


Fig. 6 The parameter effect on the surface coverage profile with a spherical shape factor and slab shape factor. Panel (a) and (b) present the effect of tortuosity per porosity, (c) and (d) show the reactant molar mass variation, (e) and (f) show the reactant areal number density (atoms per nm^2), (g) and (h) show a varying $\bar{s} = A_0/V_p$ (m^{-1}), (i) and (j) show the sticking probability β_0 , while (k) and (l) show the pore diameter (nm). The parameter effect is addressed by changing one parameter at a time and all other values are baseline parameters with a 500 s duration reactant exposure (see Tables 1 and 2). The related volumetric reactant number density profiles are presented in Fig. 5.

3.4 The effect of porous media physical characteristics on the coating penetration depth

The physical characteristics related to porous materials, such as the specific surface area, pore volume and pore size have a distinctive effect on the ALD coating penetration depth. Fig. 7 presents the simulation results for the spherical particle geometry and varied porous material properties. Different to the previously presented sensitivity analysis, three parameters are changed at the same time. Typically, with heterogeneous

catalyst supports, increasing the specific surface area (A_0) will increase the pore volume (V_p) and decrease the pore diameter (d_{pore}). The varied values range from $A_0 = 100\text{--}1000 \text{ m}^2 \text{ g}^{-1}$, $d_{\text{pore}} = 3\text{--}50 \text{ nm}$, $V_p = 0.55\text{--}0.70 \text{ cm}^3 \text{ g}^{-1}$, $\bar{s} = 1.8 \times 10^8\text{--}1.4 \times 10^9 \text{ (m}^{-1}\text{)}$. A full list of parameters and corresponding Knudsen numbers can be found from the ESI† (Table S2). Fig. 7 presents the effect of (a) the specific surface area, (b) pore diameter and (c) pore volume on the achieved penetration depth ($r_{\theta=1/2}$). All panels present results for three particle radius $R = 100$,



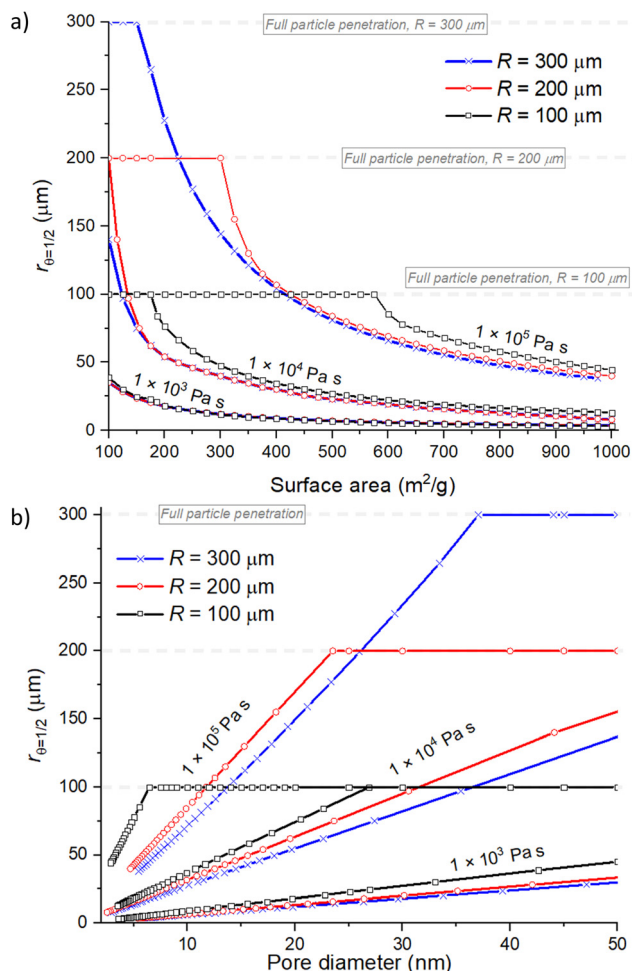


Fig. 7 Achieved penetration depth ($r_{\theta=1/2}$) with three varying parameters (a) the surface area ($\text{m}^2 \text{g}^{-1}$) and (b) pore diameter (nm). The results are shown for particle radii of 100, 200 and 300 μm and with reactant exposures of 1×10^3 , 1×10^4 and 1×10^5 Pa s at 423 K. Other parameters are baseline parameters and the used shape factor spherical particle geometry. Values used for $A_{\text{O}} = 100\text{--}1000 \text{ m}^2 \text{g}^{-1}$ and $d_{\text{pore}} = 3\text{--}50 \text{ nm}$. All varied parameters and values for the Knudsen number are presented in the ESI† Table S2.

200 and 300 μm . The simulations were conducted with three reactant exposures: 1×10^3 , 1×10^4 and 1×10^5 Pa s at 423 K.

In Fig. 7(a), the coating penetration depth decreases when the specific surface area (A_{O}) increases. The increasing specific surface area increases the number of adsorption sites per unit volume, resulting in a reduced penetration depth. The pore diameter (d_{pore}) in Fig. 7(b) is included in the Knudsen diffusion coefficient (D_{Kn}) (see eqn (5)). Therefore, the pore diameter has significant effect on the reactant travel inside the particle through the Knudsen diffusion coefficient. Increasing the pore diameter increases the Knudsen diffusion coefficient (D_{Kn}), which will increase the achieved penetration depth *via* faster reactant diffusion inside the porous media.

3.5 Temperature effect on the reactant exposure and transport inside porous media

With diffusion–reaction models presented in the literature,^{1,31,32} the reactant exposure (Pa s) is determined from the partial pressure multiplied by the reactant exposure time. In these previous studies, at least Yim *et al.*³¹ and Gayle *et al.*¹ have reported that increasing the deposition temperature reduces the achieved ALD coating penetration depth. Fig. 8 presents simulation results for the effect of varying the temperature on the volumetric reactant number density (n_{P0}) profiles and achieved coating penetration depths.

Our diffusion–reaction model uses an absolute number of reactant molecules (volumetric reactant number density, n_{P0}) to represent the initial reactant quantity at $t = 0$. Fig. 8(a) illustrates the simulation results with a fixed initial volumetric reactant number density ($1.94 \times 10^{22} \text{ m}^{-3}$). Here, the absolute quantity of reactant molecules is fixed and the reactant partial pressure increases with the increasing temperature. As the absolute reactant quantity remains unchanged, the penetration depth increases due to temperature assisted diffusion and the increasing rate of surface reactions. As the sticking probability (β_0) remains unchanged, the increased rate of surface reactions is related to the increased frequency of reactant collisions with the surface.

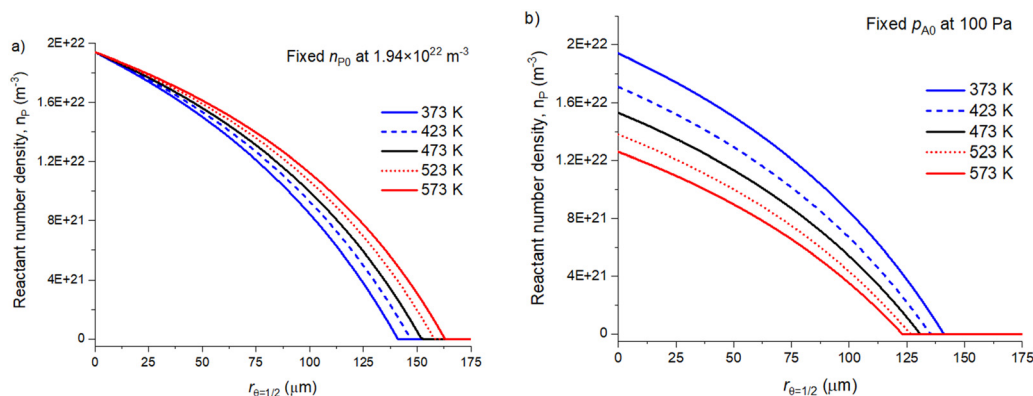


Fig. 8 The effect of selecting the volumetric reactant number density (m^{-3}) or partial pressure (Pa) in a diffusion–reaction model as a function of temperature. Panel (a) shows the fixed volumetric reactant number density to $1.94 \times 10^{22} \text{ m}^{-3}$ and (b) shows the fixed partial pressure to 100 Pa as a function of the achieved penetration depth ($r_{\theta=1/2}$). A full list of the simulation parameters and plotted values are presented in the ESI† in Table S4.



If a partial pressure (p_{A0}) would be selected for the initial reactant quantity, this value has a temperature dependency. Fig. 8(b) illustrates profiles when volumetric reactant number density is determined from a fixed 100 Pa partial pressure (p_{A0}) at varying temperatures. For example, at a temperature of 373 K, a 100 Pa partial pressure corresponds to n_{p0} of $1.94 \times 10^{22} \text{ m}^{-3}$ and at a temperature of 573 K, a 100 Pa partial pressure corresponds to $1.26 \times 10^{22} \text{ m}^{-3}$ volumetric reactant number density. Here, the achieved penetration depth decreases as a function of temperature. The decrease is due to the decreasing initial volumetric reactant number density.

4. Conclusions

This work presents a diffusion–reaction model for ALD on porous particles with three different particle geometries: a sphere, cylinder and slab. The particle geometries are presented by means of shape factors (symbol s) for sphere ($s = 2$), cylinder ($s = 1$) and slab ($s = 0$). This work shows that within a spherical porous particle, the reactant diffusion propagates from three radial directions towards the particle centre (three diffusion propagation dimensions) and the decreasing volume towards the centre of a spherical particle results in decreased reactant consumption. Compared to the slab geometry, less reactant is needed to coat spherical and cylindrical particles.

By comparing the model results for the slab and spherical particle geometries, we found that when the reactant penetration depth ($r_{\theta=1/2}$) is less than 5% of the particle radius (R), all shape factors result in similar estimates (difference within 1%) for the penetration depth. However, when increasing the penetration depth with respect to the particle radius, the difference between shape factors increases. With the slab shape factor, the penetration depth increases linearly as a function of the square root of the reactant exposure. However, this relationship is different for the spherical particle, where the shell volume available for reactant deposition decreases towards the spherical particle centre as r^3 , where r is the radial distance from the particle surface.

In addition to the analysis of the shape factors, the presented diffusion–reaction model was used to perform a parameter sensitivity analysis to address the parameter effects on the model results. The varied parameters included porosity (ε), tortuosity (τ), reactant molar mass, areal number density of metal atoms deposited on each exposure cycle upon saturation (σ), the specific surface area (A_O), pore volume (V_P), sticking probability (β_0), and pore diameter (d_{pore}). Furthermore, the single parameter variation was extended to address in detail the combined effect of the specific surface area (A_O), mean pore diameter (d_{pore}) and pore volume (V_P) on the achieved penetration depth. The general trend was that with a fixed exposure, increasing the specific surface area decreases the coating penetration depth, while increasing the pore diameter decreases the coating penetration depth, and increasing the pore volume increases the coating penetration depth. Lastly, we conceptually compared the effect of the ALD process

temperature on the diffusion–reaction model volumetric reactant number density (m^{-3}) and partial pressure (Pa). Our diffusion–reaction model uses the volumetric reactant number density (n_{p0}) to represent the initial reactant quantity. Here, the absolute initial reactant quantity has no temperature dependency, whereas the initial reactant quantity given as the partial pressure (p_{A0}) depends on the temperature. Then, the absolute quantity of initial reactant molecules and the achieved penetration depth decrease as a function of the increasing temperature.

This work presents a diffusion–reaction modelling approach to ALD on porous media with different particle shapes. We used a porous and tortuous heterogeneous catalyst particle as an example substrate. Typically, catalyst particles are simplified as a spherical particle ($s = 2$). Furthermore, the model can be implemented for the ALD process on many other shapes of porous media, for example, porous membranes ($s = 0$), high aspect ratio fibres ($s = 1$) and monoliths ($s = 1$). The selection of the particle shape factor and corresponding partial differential equation becomes increasingly relevant as the coating penetration depth is approaching the particle centre.

Symbols and abbreviations

ALD	Atomic layer deposition
a	Shape factor; $a = s + 1$ (–)
A_O	Specific surface area ($\text{m}^2 \text{g}^{-1}$)
A_{Par}	Catalyst particle outer surface area, ideal non-porous sphere (m^2)
β_0	Initial sticking probability (–)
D_e	Effective diffusion coefficient ($\text{m}^2 \text{s}^{-1}$)
D_i	Gas phase diffusion coefficient ($\text{m}^2 \text{s}^{-1}$)
D_{Kn}	Knudsen diffusion coefficient ($\text{m}^2 \text{s}^{-1}$)
D_{mi}	Molecular diffusion coefficient ($\text{m}^2 \text{s}^{-1}$)
d_{pore}	Mean pore diameter (m)
v_{th}	Mean thermal velocity (m s^{-1})
ε	Porosity ($0 \leq \varepsilon \leq 1$)
GPC	Growth-per-cycle
h	Height (m)
k_B	Boltzmann constant ($\text{m}^2 \text{kg s}^{-2} \text{K}^{-1}$)
m_r	Mass for a single reactant molecule (g)
n_{p0}	Volumetric reactant number density (m^{-3})
p_{A0}	Partial pressure of component A (Pa)
r	Radial coordinate (m)
$r_{\theta=1/2}$	ALD reactant penetration depth (μm)
R	Particle radius (m)
s	Shape factor for a particle; $s = a - 1$ (–)
\bar{s}	Ratio of surface area (A_O) and pore volume (V_P)
s_0	Average surface area of adsorption sites (m^2)
SCM	Shrinking core model
σ	Areal number density of metal atoms deposited on each exposure cycle upon saturation (m^{-2})
t	Time (s)
τ	Tortuosity (–)
T	Temperature (K)



θ	Surface coverage ($0 \leq \theta \leq 1$)
V_p	Pore volume ($\text{m}^3 \text{g}^{-1}$)
V_{Par}	Catalyst particle volume (m^3)

Author contributions

N. H. was the main contributor to interpret results and prepare the manuscript. He also finalized the diffusion–reaction model Matlab code with J. L., who also wrote the original diffusion–reaction model Matlab code and assisted N. H. in the implementation. R. L. P. suggested the initial idea to adapt particle geometries into an ALD diffusion–reaction model and supervised the manuscript writing process.

Conflicts of interest

The authors declare no conflict of interest.

Acknowledgements

The work was financially supported by Business Finland project Forest CUMP (Dnro 2158/31/2022), Business Finland project e-Fuel (Dnro 43287/31/2020) and the Academy of Finland ALDI consortium (decision no. 331082). The authors would like to acknowledge Jihong Yim for valuable discussions and kind assistance during the manuscript preparation.

References

- 1 A. J. Gayle, Z. J. Berquist, Y. Chen, A. J. Hill, J. Y. Hoffman, A. R. Bielinski, A. Lenert and N. P. Dasgupta, Tunable Atomic Layer Deposition into Ultra-High-Aspect-Ratio ($>60000:1$) Aerogel Monoliths Enabled by Transport Modeling, *Chem. Mater.*, 2021, **33**, 5572–5583.
- 2 A. W. Ott, K. C. McCarley, J. W. Klaus, J. D. Way and S. M. George, Atomic layer controlled deposition of Al_2O_3 films using binary reaction sequence chemistry, *Appl. Surf. Sci.*, 1996, **107**, 128–136.
- 3 T. Keuter, G. Mauer, F. Vondahlen, R. Iskandar, N. H. Menzler and R. Vaßen, Atomic-layer-controlled deposition of TEMAZ/ O_2 -ZrO₂ oxidation resistance inner surface coatings for solid oxide fuel cells, *Surf. Coat. Technol.*, 2016, **288**, 211–220.
- 4 W. Szmyt, C. Guerra-Nuñez, L. Huber, C. Dransfeld and I. Utke, Atomic Layer Deposition on Porous Substrates: From General Formulation to Fibrous Substrates and Scaling Laws, *Chem. Mater.*, 2022, **34**, 203–216.
- 5 B. J. O'Neill, D. H. K. K. Jackson, J. Lee, C. Canlas, P. C. Stair, C. L. Marshall, J. W. Elam, T. F. Kuech, J. A. Dumesic, G. W. Huber, B. J. Oneill, D. H. K. K. Jackson, J. Lee, C. Canlas, P. C. Stair, C. L. Marshall, J. W. Elam, T. F. Kuech, J. A. Dumesic and G. W. Huber, Catalyst design with atomic layer deposition, *ACS Catal.*, 2015, **5**, 1804–1825.
- 6 J. A. Libera, J. W. Elam and M. J. Pellin, Conformal ZnO coatings on high surface area silica gel using atomic layer deposition, *Thin Solid Films*, 2008, **516**, 6158–6166.
- 7 J. R. van Ommen, A. Goulas and R. L. Puurunen, *Kirk-Othmer Encyclopedia of Chemical Technology*, Wiley, 2021, vol. 51, pp. 1–42.
- 8 S. M. George, Atomic layer deposition: An overview, *Chem. Rev.*, 2010, **110**, 111–131.
- 9 T. Suntola, Atomic layer epitaxy, *Mater. Sci. Rep.*, 1989, **4**, 261–312.
- 10 E. L. Lakomaa, Atomic layer epitaxy (ALE) on porous substrates, *Appl. Surf. Sci.*, 1994, **75**, 185–196.
- 11 E. Sarnello, Z. Lu, S. Seifert, R. E. Winans and T. Li, Design and Characterization of ALD-Based Overcoats for Supported Metal Nanoparticle Catalysts, *ACS Catal.*, 2021, **11**, 2605–2619.
- 12 J. A. Singh, N. Yang and S. F. Bent, Nanoengineering Heterogeneous Catalysts by Atomic Layer Deposition, *Annu. Rev. Chem. Biomol. Eng.*, 2017, **8**, 41–62.
- 13 J. Lu, B. Liu, J. P. Greeley, Z. Feng, J. A. Libera, Y. Lei, M. J. Bedzyk, P. C. Stair and J. W. Elam, Porous alumina protective coatings on palladium nanoparticles by self-poisoned atomic layer deposition, *Chem. Mater.*, 2012, **24**, 2047–2055.
- 14 M. Lindblad, S. Haukka, A. Kytökivi, E.-L. Lakomaa, A. Rautiainen and T. Suntola, Processing of catalysts by atomic layer epitaxy: modification of supports, *Appl. Surf. Sci.*, 1997, **121–122**, 286–291.
- 15 R. K. Ramachandran, C. Detavernier and J. Dendooven, in *Nanotechnology in Catalysis: Applications in the Chemical Industry, Energy Development, and Environment Protection*, ed. B. F. Sels and M. Van de Voorde, John Wiley & Sons, Incorporated, Vienna, 2017, p. 1183.
- 16 A. Arandia, J. Yim, H. Warraich, E. Leppäkangas, R. Bes, A. Lempelto, L. Gell, H. Jiang, K. Meinander, T. Viinikainen, S. Huotari, K. Honkala and R. L. Puurunen, Effect of atomic layer deposited zinc promoter on the activity of copper-on-zirconia catalysts in the hydrogenation of carbon dioxide to methanol, *Appl. Catal., B*, 2023, **321**, 122046.
- 17 H. Zhou, S. R. Docherty, N. Phongprueksathat, Z. Chen, A. V. Bukhtiyarov, I. P. Prosvirin, O. V. Safonova, A. Urakawa, C. Copéret, C. R. Müller and A. Fedorov, Combining Atomic Layer Deposition with Surface Organometallic Chemistry to Enhance Atomic-Scale Interactions and Improve the Activity and Selectivity of Cu–Zn/SiO₂ Catalysts for the Hydrogenation of CO₂ to Methanol, *JACS Au*, 2023, **3**, 2536–2549.
- 18 R. G. R. G. Gordon, D. Hausmann, E. Kim and J. Shepard, A kinetic model for step coverage by atomic layer deposition in narrow holes or trenches, *Chem. Vap. Deposition*, 2003, **9**, 73–78.
- 19 J. Dendooven, D. Deduytsche, J. Musschoot, R. L. Vanmeirhaeghe and C. Detavernier, Modeling the Conformality of Atomic Layer Deposition: The Effect of Sticking Probability, *J. Electrochem. Soc.*, 2009, **156**, P63.
- 20 N. Yazdani, V. Chawla, E. Edwards, V. Wood, H. G. Park and I. Utke, Modeling and optimization of atomic layer



- deposition processes on vertically aligned carbon nanotubes, *Beilstein J. Nanotechnol.*, 2014, 5, 234–244.
- 21 V. Cremers, F. Geenen, C. Detavernier and J. Dendooven, Monte Carlo simulations of atomic layer deposition on 3D large surface area structures: Required precursor exposure for pillar- versus hole-type structures, *J. Vac. Sci. Technol., A*, 2017, 35, 01B115.
 - 22 V. Cremers, R. L. Puurunen and J. Dendooven, Conformality in atomic layer deposition: Current status overview of analysis and modelling, *Appl. Phys. Rev.*, 2019, 6, 021302.
 - 23 A. Yanguas-Gil and J. W. Elam, A Markov chain approach to simulate Atomic Layer Deposition chemistry and transport inside nanostructured substrates, *Theor. Chem. Acc.*, 2014, 133, 1465.
 - 24 R. A. Adomaitis, A ballistic transport and surface reaction model for simulating atomic layer deposition processes in high-aspect-ratio nanopores, *Chem. Vap. Deposition*, 2011, 17, 353–365.
 - 25 P. Poodt, A. Mameli, J. Schulpen, W. M. M. (Erwin) Kessels and F. Roozeboom, Effect of reactor pressure on the conformal coating inside porous substrates by atomic layer deposition, *J. Vac. Sci. Technol., A*, 2017, 35, 021502.
 - 26 M. C. Schwille, J. Barth, T. Schössler, F. Schön, J. W. Bartha and M. Oettel, Simulation approach of atomic layer deposition in large 3D structures, *Modell. Simul. Mater. Sci. Eng.*, 2017, 25, 035008.
 - 27 J. W. Elam, D. Routkevitch, P. P. Mardilovich and S. M. George, Conformal Coating on Ultrahigh-Aspect-Ratio Nanopores of Anodic Alumina by Atomic Layer Deposition, *Chem. Mater.*, 2003, 15, 3507–3517.
 - 28 H. C. M. Knoops, E. Langereis, M. C. M. van de Sanden and W. M. M. Kessels, Conformality of Plasma-Assisted ALD: Physical Processes and Modeling, *J. Electrochem. Soc.*, 2010, 157, G241.
 - 29 N. Heikkinen, J. Lehtonen, L. Keskiäli, J. Yim, S. Shetty, Y. Ge, M. Reinikainen and M. Putkonen, Modelling atomic layer deposition overcoating formation on a porous heterogeneous catalyst, *Phys. Chem. Chem. Phys.*, 2022, 24, 20506–20516.
 - 30 A. Yanguas-Gil, *Growth and Transport in Nanostructured Materials*, Springer International Publishing, Cham, 2017.
 - 31 J. Järvillehto, J. A. Velasco, J. Yim, C. Gonsalves and R. L. Puurunen, Simulation of conformality of ALD growth inside lateral channels: comparison between a diffusion-reaction model and a ballistic transport-reaction model, *Phys. Chem. Chem. Phys.*, 2023, 25, 22952–22964.
 - 32 M. Ylilampi, O. M. E. Ylivaara and R. L. Puurunen, Modeling growth kinetics of thin films made by atomic layer deposition in lateral high-aspect-ratio structures, *J. Appl. Phys.*, 2018, 123, 205301.
 - 33 A. Yanguas-Gil and J. W. Elam, Self-limited reaction-diffusion in nanostructured substrates: Surface coverage dynamics and analytic approximations to ALD saturation times, *Chem. Vap. Deposition*, 2012, 18, 46–52.
 - 34 T. Keuter, N. H. Menzler, G. Mauer, F. Vondahlen, R. Vaßen and H. P. Buchkremer, Modeling precursor diffusion and reaction of atomic layer deposition in porous structures, *J. Vac. Sci. Technol., A*, 2015, 33, 01A104.
 - 35 H. S. Fogler, *Elements of Chemical Reaction Engineering*, Upper Saddle River, New Jersey, USA, 3rd edn, 2004.
 - 36 T. Salmi, J. Wärnä, J. R. Hernández Carucci and C. A. de Araújo Filho, *Chemical Reaction Engineering*, De Gruyter, Berlin, 2020.
 - 37 L. D. Schmidt, *The Engineering of Chemical Reactions*, 1998, vol. 45, p. 536.
 - 38 E. Leonardi and C. Angeli, Transient Diffusion within Spherical Particles: Numerical Resolution of the Maxwell–Stefan Formulation, *Ind. Eng. Chem. Res.*, 2010, 49, 5654–5660.
 - 39 H. Y. Lee, C. J. An, S. J. Piao, D. Y. Ahn, M. T. Kim and Y. S. Min, Shrinking core model for Knudsen diffusion-limited atomic layer deposition on a nanoporous monolith with an ultrahigh aspect ratio, *J. Phys. Chem. C*, 2010, 114, 18601–18606.
 - 40 C. Detavernier, J. Dendooven, S. Pulinthanathu Sree, K. F. Ludwig and J. A. Martens, Tailoring nanoporous materials by atomic layer deposition, *Chem. Soc. Rev.*, 2011, 40, 5242.
 - 41 X. Yang, A. B. F. Martinson, J. W. Elam, L. Shao and S. B. Darling, Water treatment based on atomically engineered materials: Atomic layer deposition and beyond, *Matter*, 2021, 4, 3515–3548.
 - 42 A. C. Liapis, A. Subramanian, S. Cho, K. Kisslinger, C. Nam and S. Yun, Conformal Coating of Freestanding Particles by Vapor-Phase Infiltration, *Adv. Mater. Interfaces*, 2020, 7, 1–6.
 - 43 J. Yim, E. Verkama, J. A. Velasco, K. Arts and R. L. Puurunen, Conformality of atomic layer deposition in microchannels: impact of process parameters on the simulated thickness profile, *Phys. Chem. Chem. Phys.*, 2022, 24, 8645–8660.
 - 44 Y. Yong, X. Lou, S. Li, C. Yang and X. Yin, Direct simulation of the influence of the pore structure on the diffusion process in porous media, *Conf. Math. Appl.*, 2014, 67, 412–423.
 - 45 L. Shen and Z. Chen, Critical review of the impact of tortuosity on diffusion, *Chem. Eng. Sci.*, 2007, 62, 3748–3755.
 - 46 R. L. Puurunen, Growth per cycle in atomic layer deposition: Real application examples of a theoretical model, *Chem. Vap. Deposition*, 2003, 9, 327–332.
 - 47 B. E. Rapp, *Microfluidics: modeling mechanics and mathematics*, William Andrew, Amsterdam, Netherlands, 1st edn, 2017.
 - 48 W. Schiesser, *The Numerical Method of Lines – Integration of Partial Differential Equations*, Academic Press, 1st edn, 1991.
 - 49 R. L. Puurunen, Surface chemistry of atomic layer deposition: A case study for the trimethylaluminum/water process, *J. Appl. Phys.*, 2005, 97, 121301.
 - 50 A. D. McNaught and A. Wilkinson, *The IUPAC Compendium of Chemical Terminology*, International Union of Pure and Applied Chemistry (IUPAC), Research Triangle Park, NC, 2nd edn, 2019.
 - 51 K. Arts, V. Vandalon, R. L. Puurunen, M. Utriainen, F. Gao, W. M. M. (Erwin) Kessels and H. C. M. Knoops, Sticking probabilities of H₂O and Al(CH₃)₃ during atomic layer deposition of Al₂O₃ extracted from their impact on film conformality, *J. Vac. Sci. Technol., A*, 2019, 37, 030908.

