




 Cite this: *Phys. Chem. Chem. Phys.*, 2024, 26, 7407

Reactivity of cationic silver clusters with O₂: a probe of interplay between clusters' geometric and electronic structures†

 Jin Hu, Jun Ma, Zhengqian Jin, Wen Liu, Lulu Huang, Xuefeng Wang * and Xiaopeng Xing *

We explored the size-dependent reactivity of Ag_n⁺ (*n* = 2–22) with O₂ under mild conditions and found that only a few sizes of Ag_n⁺, with even values of *n* = 4, 6, 12, 16, 18, and 22, are reactive. Possible structures of Ag_n⁺ (*n* = 2–22) were determined using a genetic algorithm with incomplete local optimizations at the DFT level, and the calculated bonding strengths of O₂ on these structures are consistent with experimental observations. Analyses revealed a close relationship between the reactivity of Ag_n⁺ with O₂ and its HOMO–LUMO gap: cationic silver clusters with a small HOMO–LUMO gap are reactive, which can be rationalized by the covalent character of chemical bonds between Ag_n⁺ and O₂ involving their frontier orbitals. The peculiar size-dependent HOMO–LUMO gaps and reactivity with O₂ correlate with the subtle interplay between the electronic configurations and geometric structures of these silver cluster cations.

 Received 19th October 2023,
 Accepted 26th January 2024

DOI: 10.1039/d3cp05082c

rsc.li/pccp

1. Introduction

The geometric and electronic structures of small clusters are fundamental to understanding their properties and exploring their potential applications.^{1–4} For a bulk system, the geometric and electronic structures are independent, while the electronic properties of nanoparticles or sub-nano clusters are generally correlated with their geometric structures. Like small molecules, the geometric and electronic structures of sub-nano clusters interplay with each other, which can be rationalized using the Jahn–Teller effect.^{5–9} Metal clusters with good ductility and relatively simple electron configurations are ideal examples to explore this interplay, and the free electron shell model applied in these species needs to take their geometric characteristics into account.^{10–12}

Silver clusters serve as valuable model systems for investigating the active sites of silver-based catalysts in various

oxidation processes with significant industrial relevance, for example, the partial oxidation of ethylene to ethylene oxide,^{13,14} partial oxidation of methanol or ethanol to their aldehydes,¹⁵ and selective oxidation removal of CO from H₂ used in fuel cells.^{16,17} In addition, the relatively simple electronic configuration and bonding character of the silver element make silver clusters ideal models for understanding many fundamental concepts in the cluster field. By analyzing the mass spectra of Ag_n⁺ and Ag_n formed in the gas phase, previous investigations showed that the enhanced abundances of certain sizes can be rationalized using the free electron model initially observed in the alkali metal clusters.^{18,19} The electronic properties of Ag_n[−] were characterized using photoelectron spectroscopy, and size-dependent electron detachment energies provide evidence for the applicability of the electron shell model in anionic silver systems.^{20–22} The geometric structures of some Ag_n[−] and Ag_n⁺ species were explored by combining the fingerprint information from various experiments using photoelectron spectroscopy, ion mobility, and trapped ion electron diffraction techniques and theoretical calculations.^{21–27}

Reactions between Ag_n[−] and O₂ used as supplemental probes for the electronic properties of anionic silver clusters according to a scenario in which one electron transfers from the anionic silver moiety to the adsorbed O₂ in formed complexes.^{28–33} There are drastic changes in the reactivity for sizes of Ag_n[−] containing 8 and 20 free electrons, which implies the formation of electron shells like those in an atom.^{30,31} The interplay between the geometric and electronic structures was

School of Chemical Science and Engineering, Shanghai Key Lab of Chemical Assessment and Sustainability, Tongji University, 1239 Siping Road, Shanghai, 200092, China. E-mail: xfwang@tongji.edu.cn, xingxp@tongji.edu.cn

† Electronic supplementary information (ESI) available: The mass spectra of Ag_n⁺ (*n* = 1–3) with the presence of O₂ (Fig. S1; PDF); the calculated low-lying structures of Ag_n⁺ (*n* = 2–22) (Fig. S2–S10; PDF); the reaction paths of the dominant structures of Ag_n⁺ (*n* = 4, 6, 8, 10, 12) with O₂, the lowest-lying structures of Ag_nO₂⁺ (*n* = 3–12) with O₂ inserted into or dissociated on the silver moiety, the relative energies of multiple products of Ag₁₀⁺ with O₂, and the bonding scheme of Ag₄O₂⁺ and Ag₆O₂⁺ (Fig. S11–S14; PDF); the schematic experimental setup (Fig. S15; PDF). See DOI: <https://doi.org/10.1039/d3cp05082c>



discussed in sizes containing 14 and 30 free electrons.^{30,33} A recent study combining reaction experiments and theoretical calculations showed the remarkable stability of Ag_{17}^- due to its closed electronic as well as geometric shells,³² and a single congener doping atom can result in the rearrangement of the geometry of Ag_{17}^- and significantly change its electronic properties and reactivity.³⁴

In contrast to the good understanding of the electronic properties of anionic silver clusters and therefore the correlation with the reactivity with O_2 , there are no efficient techniques to directly characterize the electronic properties of cationic silver clusters. Reactions between Ag_n^+ and O_2 species were previously explored in the extension region of the cluster source at low temperatures.^{35–39} The results showed a general even–odd oscillation of the reactivity in the size range up to one nanometre. It was proposed that Ag_n^+ containing 2, 8, and 20 free electrons form closed electron shells, which endows them enhanced stability, and a slight increase of the temperature (~ 105 K) leads to the dissociation of O_2 on Ag_n^+ to form the oxide state of silver.^{35,38,39} A recent experiment using an ion trap running at room temperature showed that the even-odd oscillation in the reactivity of Ag_n^+ ends at $n = 10$, and all larger sizes are highly reactive.⁴⁰ In these previous experiments, cluster etching paths and reactive odd sizes of Ag_n^+ are hints that extra energies could be ubiquitous in reaction systems, facilitating the crossing of barriers associated with bond breakages or spin accommodation requirements. The reactivity of clusters under relatively violent conditions could partially mask the effects of their intrinsic geometric or electronic characteristics.

Herein, we explored the reactions of Ag_n^+ ($n = 2–22$) with O_2 under mild conditions, in which clusters and reactant O_2 were pre-cooled by a helium buffer gas before entering the reaction cell, it was ensured that reaction systems were canonical ensembles at a defined low temperature.^{31,41} Under conditions described in the Experimental section, a peculiar discontinuous even–odd oscillation was observed for the of reactivity. In addition to the inertness of odd sizes which was attributed to barriers from the spin accommodation requirement,³⁰ a few even sizes, including Ag_4^+ and Ag_{10}^+ which were expected to have alkali-like electron configurations and high reactivity,^{36,39} were shown to be inert. Combining these carefully designed experiments and large theoretical calculations, we found that the peculiar discontinuous odd–even oscillation of the reactivity of clusters reflects the size-dependent feasibility of the transition from physisorption to chemisorption of the combined O_2 . This feasibility closely correlates with the varying HOMO–LUMO gaps of Ag_n^+ , which reflect a subtle interplay between their electronic and geometric structures.

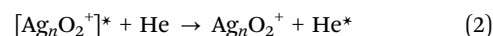
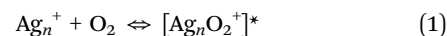
2. Experimental and computational methods

The cluster reaction experiments were carried out using an instrument composed of a magnetron sputtering cluster source, a continuous micro tube reactor and a time-of-flight

(TOF) mass spectrometer. The device diagram is shown in Fig. S15 (ESI[†]), and the details of this instrument were described elsewhere.^{31,42} Briefly, the cluster source has a liquid-nitrogen-cooled aggregation chamber enclosing a magnetron sputter discharge head. A small tube reactor was installed at the exit of the source chamber, which is isolated from the source by a ceramic spacer. The temperature of this tube reactor was adjusted using a controlled heater and was stabilized at 150 K in this work.

The clusters formed inside the source chamber were carried by the helium buffer gas and argon discharging gas (110 sccm and 14 sccm, respectively) and flowed into the tube reactor. Only those species near the central axis can enter the reaction region, and most clusters were filtered by multi-layer meshes. The intentional decrease in the concentration of metal species in the reaction region is to stop the cluster growth processes. Another function of the multi-layer meshes is to thermalize the carrier gas to the set temperature of the reactor. The reactant O_2 was introduced through two copper tubes extending close to the axial center of the reactor, which were also thermalized to the temperature of the reactor inside the mantle layer. The flow rates were about 0.02–0.2 sccm. The total pressure inside the aggregation chamber of the cluster source and tube reactor was about 0.5 torr, and the partial pressure of the reactant in the reactor was around 10^{-4} torr. The reaction time was estimated to be several milliseconds. The generated species, including parent clusters and reaction products, went through a skimmer at the end of the flow reactor. The continuous ion beam from the skimmer was directed to and analyzed using a time-of-flight (TOF) mass spectrometer operating at 800 Hz. The mass spectra of species at a series of O_2 flow rates were recorded using the TOF mass spectrometer. The intensity of one parent cluster Ag_n^+ at a certain O_2 flow rate (I) and that without O_2 (I_0) were separately integrated.

The adsorption process of clusters and O_2 molecules follows the Lindemann reaction mechanism. Taking Ag_n^+ and O_2 as an example:



and the reaction rate can be written in the following form:

$$\frac{dI}{dt} = k_{\text{III}}[\text{O}_2][\text{He}]I \quad (3)$$

Integrate both sides to get:

$$-\ln \frac{I}{I_0} = k_{\text{III}}[\text{O}_2][\text{He}]t \quad (4)$$

The rate constant k_{III} is proportional to $-\ln(I/I_0)/n$ since true concentrations of O_2 and He (shown as $[\text{O}_2]$ and $[\text{He}]$) and the reaction time t are constant. Besides, $[\text{O}_2]$ and $[\text{He}]$ are much larger than the partial pressure of metal clusters, so the reaction can be regarded as a quasi-first-order reaction. We plotted the obtained $-\ln(I/I_0)$ vs. the flow of O_2 , and the slope of the resulting straight line is proportional to k_{III} .



The structure search of silver clusters was conducted using our newly encoded genetic algorithm program with an incomplete local optimization strategy,³³ in which all optimization processes and energy calculations were carried out using the Gaussian09 program.⁴³ Briefly, we used a module to generate random structures including, a space-free motif, a close packing motif, a simple cubic packing motif, a cage motif, a solid sphere motif, a ring motif and other user-defined motifs. These structures were screened according to their calculated energies and evolved toward low-lying ones through multiple rounds of crossovers, mutations, and competitions. The main feature of this program is that all local optimizations in the structure evolution are terminated in a relatively flat area rather than exact local minimum points on the potential energy surface (PES), and only the structures of the last generation were completely optimized to obtain the global minimum. The efficiency and reliability of methods have been previously shown in searching the structure of Ag₃₀ and other representative clusters.³³ The initial structural candidates of Ag_nO₂⁺ reaction products were generated by randomly putting O₂ around all possible adsorption sites of obtained lowest-lying structures of Ag_n⁺, and various adsorption patterns including O-atop and O–O doubly connected geometries in their doublet or quartet state were considered. The complete and incomplete optimizations in this study were at the PBE level with the def2-SVP basis set for Ag and def2-TZVP basis sets for O.^{44–46} After each

complete optimization process, the harmonic vibrational frequencies were calculated to confirm that a real minimum point was obtained, and at the same time, the zero-point vibrational energy (ZPVE) was derived. Calculations on the relative energies of various isomers and O₂ adsorption energies were based on the electronic energies of related species modified by their ZPVE corrections. The NPA charges and spin populations on O–O in the obtained structures of Ag_nO₂⁺ were calculated.

3. Results and discussion

3.1 Size-dependent reactivity of Ag_n⁺ (*n* = 2–22) with O₂

Fig. 1a shows Ag_n⁺ (*n* = 4–23) species formed in the cluster source and their reaction products with 0.025 sccm and 0.075 sccm O₂. The vertical dashed lines indicate Ag_n⁺, and circle symbols indicate dominant products, such as Ag_nO₂⁺. For Ag_n⁺ (*n* < 4), the mass spectra contained cluster complexes with argon from the sputter source and nitrogen or CO molecules possibly from the background. However, the spectra clearly show that there were no oxygen-adsorbed products for Ag_n⁺ (*n* = 1–3) (shown in Fig. S1, ESI[†]). The relative kinetic rates of Ag_n⁺ were obtained as described in the Experimental section, and are summarized in Fig. 1b. The reactivity showed an apparent even-odd oscillation, in which all active sizes contain an even number of atoms, in other words, an odd number of valence

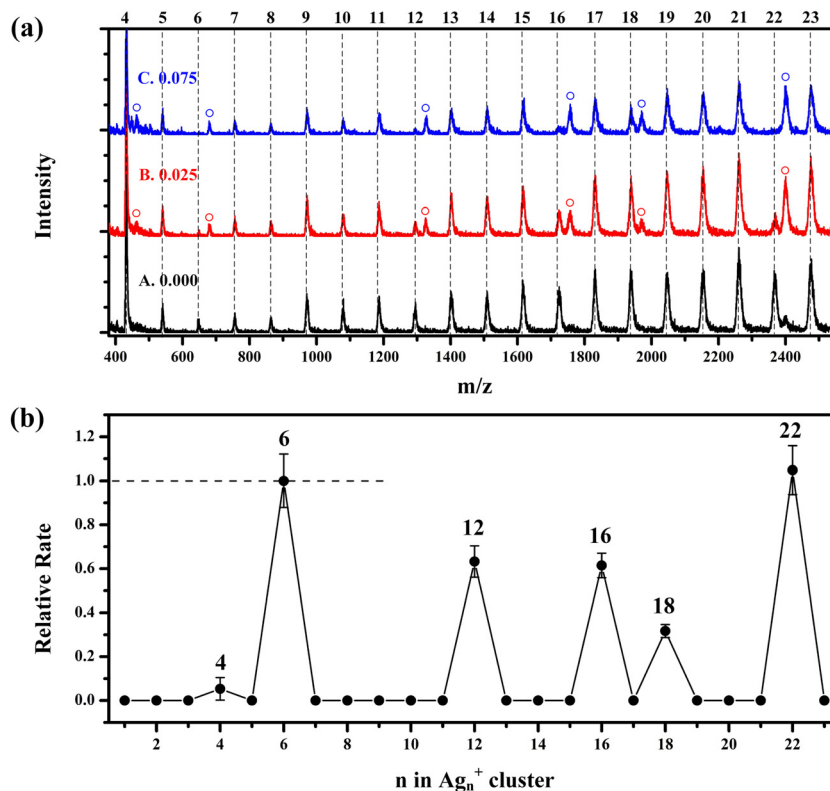


Fig. 1 (a) Mass spectra of parent and product ions for reactions between Ag_n⁺ (*n* = 4–23) and O₂ with O₂ flows of 0.000 sccm (curve A), 0.025 sccm (curve B) and 0.075 sccm (curve C) at 150 K. Dot lines in the mass spectra indicate the positions of Ag_n⁺, and circle symbols indicate dominant products, such as Ag_nO₂⁺. (b) Relative reaction rates normalized to that of Ag₆⁺. Detection limit of relative rates was estimated to be 10^{−2}.



electrons. All reactive sizes adsorbed only one O_2 . Notably, not all even sizes in this range showed detectable activity. Ag_n^+ species with $n = 8, 10, 14,$ and 20 are inert, and the reactivity of $n = 4$ is very low, which is different from previous experimental observations.^{36,39,40}

3.2 Theoretical structures and electronic properties of Ag_n^+ ($n = 2-22$)

To understand the peculiar discontinuous odd-even oscillation of the reactivity, we theoretically explored structural candidates of Ag_n^+ ($n = 2-22$) using a newly developed genetic algorithm program³³ and determined the most likely ones according to their relative energies and comparisons with the results from previous experiments. The low-lying isomers for each size of Ag_n^+ ($n = 2-22$) are displayed in Fig. S2–S10 (ESI[†]), and the most likely ones present in experiments are presented in Fig. 2a. The structures of Ag_n^+ ($n = 2-13$) were recently determined based on a combination of the far-infrared multiple photon dissociation spectra of the complexes of clusters and theoretical calculations.⁴¹ The lowest-lying structures of Ag_n^+ ($n = 2-5, 8, 10-13$) from our calculations (Fig. S2–S6, ESI[†]) are consistent with this report. The results of Ag_6^+, Ag_7^+ and Ag_9^+ in our calculations (Fig. S2–S4, ESI[†]) include

one or two structures with even lower energies at the present theoretical level, while energy differences are almost negligible. The results of Ag_n^+ ($n = 12-15$) (Fig. S5–S7, ESI[†]) are in complete agreement with those in previous theoretical studies.⁴⁷⁻⁵⁰ The minimum structure of optimized Ag_{19}^+ (Fig. S9, ESI[†]) is consistent with the one previously determined using trapped ion electron diffraction experiments and theoretical calculations.²⁷ For $n = 16, 17, 18$ and 20 (Fig. S7–S9, ESI[†]), we found one or two new structures with lower energies than those from a previous theoretical report.⁴⁹ For $n = 21$ and 22 (Fig. S10, ESI[†]), our calculation found more than seven isomers which are more stable than the lowest-lying ones reported by previous theoretical studies.^{49,50} In Fig. 2a, we list the structures of Ag_n^+ ($n = 3-13$) previously confirmed by the combined experiment and theoretical calculation⁴¹ and the lowest-lying ones of Ag_n^+ ($n = 14-22$) from our calculations. We saw that the shapes of clusters vary with increasing size: the structures undergo a 2D ($n = 4$) to 3D ($n = 5$) transition and become near-spherical at $n = 8$ and 9 ; the structures of $n = 10-14$ have elongated prolate shapes and those of $n = 15-17$ display an oblate sheet formed by two layers of staggered atoms; for sizes with $n = 17-19$, geometries change into an irregular

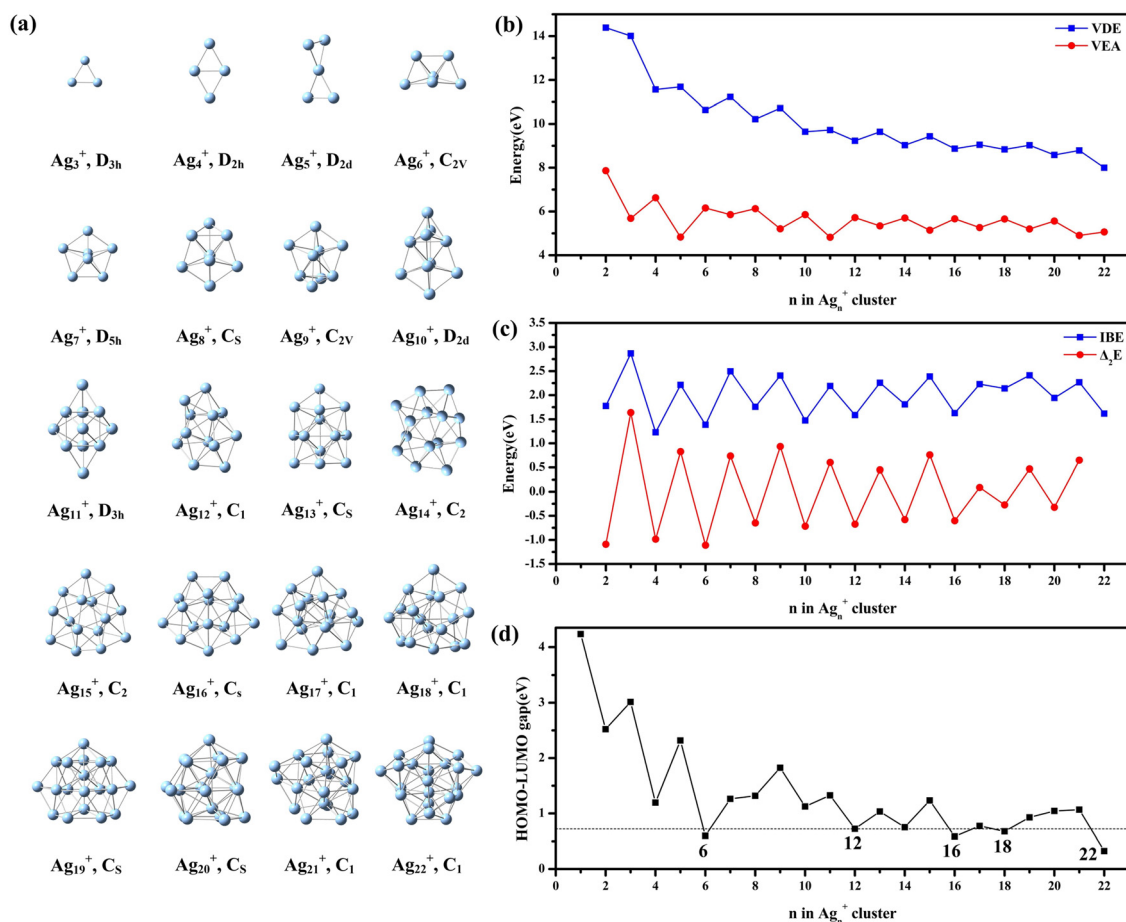


Fig. 2 (a) Structures of Ag_n^+ ($n = 3-22$) predicted by calculations at the PBE/def2-SVP level of theory, and their theoretical properties including (b) vertical detachment energies (VDE) and vertical electron affinities (VEA), (c) incremental binding energies (IBE) and the second-order difference in binding energies (Δ_2E), and (d) HOMO-LUMO (H-L) gaps. Structures of Ag_n^+ ($n = 3-13$) were determined by comparisons between low-lying structures and the available experimental information. Structures of Ag_n^+ ($n = 14-22$) were proposed to be the lowest-lying ones in calculations.



oblate shape with an embedded silver atom; for those with $n = 20$ – 22 , all atoms are tightly packed into a quasi-spherical shape and the number of silver atoms trapped inside increases to two or three.

We calculated the electronic and energetic features of the most likely structures shown in Fig. 2a, including vertical electron detachment energies (VDE), vertical electron affinities (VEA), the incremental binding energies (IBE), and second-order differences in binding energy (Δ_2E), which were defined as follows:

$$\text{VDE}(\text{Ag}_n^+) = E[\text{Ag}_n^{2+}] - E[\text{Ag}_n^+] \quad (5)$$

$$\text{VEA}(\text{Ag}_n^+) = E[\text{Ag}_n^+] - E[\text{Ag}_n^0] \quad (6)$$

$$\text{IBE}(\text{Ag}_n^+) = E[\text{Ag}^0] + E[\text{Ag}_{n-1}^+] - E[\text{Ag}_n^+] \quad (7)$$

$$\Delta_2E(\text{Ag}_n^+) = E[\text{Ag}_{n+1}^+] + E[\text{Ag}_{n-1}^+] - 2E[\text{Ag}_n^+] \quad (8)$$

As seen in Fig. 2b, there is a general odd–even oscillation in VDE and VEA. For VDE, a high value of odd-sized Ag_n^+ species is due to their lower-energy singlet state formed by an even number of electrons, compared to its even-size neighbors with one unpaired electron. Note that with an increase of n , the variation of VDE gradually becomes smooth, but significantly decreases at Ag_{22}^+ (0.58 eV lower than Ag_{20}^+ and 0.78 eV lower than Ag_{21}^+). VEA is reminiscent of a mirror image of VDE, with a large VDE corresponding to a small VEA. Comparing Fig. 1 and Fig. 2b, we see no evident correlation between the reactivity of Ag_n^+ species with oxygen and their VDEs or VEAs, which is different from the case observed for silver cluster anions.^{30–32}

IBE and Δ_2E are depicted in Fig. 2c, both of which show oscillatory trends and do not have a correlation with the reactivity in Fig. 1. HOMO–LUMO gaps in Fig. 2d exhibit a large drop from Ag_3^+ to Ag_{22}^+ . Additionally, it was found that there is a threshold of about 0.73 eV that separates reactive clusters from inert ones observed in Fig. 1. Especially, the significantly lower HOMO–LUMO gap and VDE of Ag_{22}^+ than those other species correspond to a greater propensity to adsorb one O_2 in experiments, whereas Ag_{20}^+ , with an open-shell size, unexpectedly shows a sizable gap, even larger than its neighbor Ag_{19}^+ , with a closed 18-electron shell.

3.3 Theoretical exploration of the bonding of O_2 on Ag_n^+ ($n = 2$ – 22)

We theoretically explored the structures of Ag_nO_2^+ ($n = 3$ – 12 , 20 , and 22) and bonding strengths of adsorbed O_2 . The structure searching started from the structures of Ag_n^+ ($n = 3$ – 22) shown in Fig. 2, and one O_2 unit was put on each adsorption site in an end-on or side-on manner. Then, all convergent structures were sequenced according to their energies. For all even- and odd-sized clusters, calculations located structures with an end-on O_2 . There are also some structures for even sizes with a side-on O_2 , while no such ones for odd sizes. The lowest-lying structures of Ag_nO_2^+ ($n = 3$ – 12 , 20 , and 22) with an end-on O_2 and those of the Ag_nO_2^+ ($n = 4$, 6 , 8 , 10 , 12 , 20 , and 22) with a side-on O_2 are shown in Fig. 3a and b, respectively. For nearly all structures with either an end-on or a side-on O_2 , silver moieties

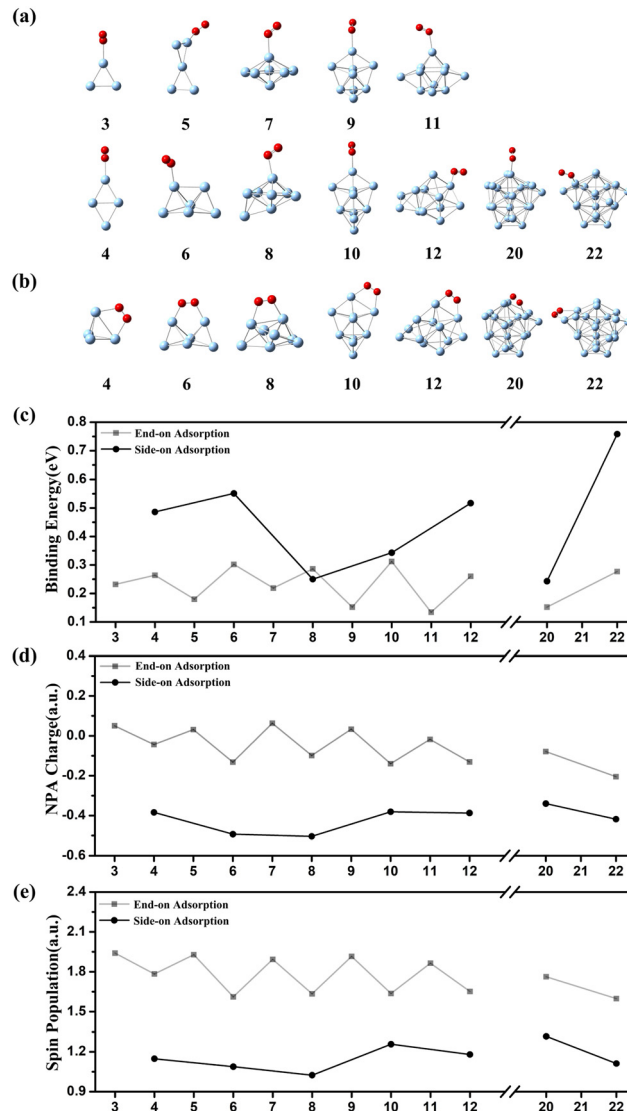


Fig. 3 (a) Structures of Ag_nO_2^+ ($n = 3$ – 12 , 20 , and 22) containing an end-on O_2 and (b) Ag_nO_2^+ ($n = 4$, 6 , 8 , 10 , 12 , 20 , and 22) containing a side-on O_2 formed by Ag_n^+ structures shown in Fig. 2; the calculated properties of above structures including (c) the binding energy of O_2 , (d) NPA charges on O_2 , and (e) spin populations on O_2 . All structures and their properties are from calculations at the PBE level with the def2-SVP basis set for Ag and def2-TZVP basis set for O.

remain the original geometries of Ag_n^+ . The only exception is the lowest-lying Ag_4O_2^+ in Fig. 3b, in which the frame of Ag_4^+ changes from a rhomboid to a pyramid after combining with a side-on O_2 . The binding energy of O_2 in each structure and charge and spin populations localized on the O_2 unit are shown in Fig. 3c–e. In Fig. 3c, the bonding energies of side-on O_2 on Ag_n^+ ($n = 4$, 6 , 12 , and 22) are higher than 0.4 eV and those of side-on O_2 on Ag_n^+ ($n = 8$, 10 , and 20) are lower than 0.35 eV; the bonding energies of end-on O_2 on all even or odd sizes are lower than 0.35 eV. These calculated binding energy values are generally consistent with experimental observations in Fig. 1, since the threshold of the binding energy to resist the entropy decrease accompanying the adsorption was estimated to be



around 0.37 eV under similar experimental condition.⁵¹ In Fig. 3d, there are very small amounts of negative charge (<0.2 a.u.) populated on end-on O₂, while the amounts on side-on O₂ in even sizes are apparently larger. In Fig. 3e, the spin populations on side-on O₂ are 1.6–2.0 a.u., while those on end-on O₂ are 1.0–1.3 a.u. These apparently different charge and spin values imply that end-on and side-on O₂ are in different states. To summarize the findings from bonding energies and charge and spin populations, end-on O₂ units are physisorbed and side-on O₂ units found in even sizes are chemically bonded. The transition paths from the end-on state to the side-on state of O₂ in Ag_{*n*}⁺ (*n* = 4, 6, 8, 10, and 12) were theoretically explored, and the results are shown in Fig. S11a–e (ESI†). It is interesting to find that the variation of the kinetic barriers of the transitions of these five sizes is consistent with the changes of their HOMO–LUMO gaps, which are summarized in Fig. S11f (ESI†). These trends are also consistent with the observed reactivity variation shown in Fig. 1. Especially, the calculated high binding energy (0.48 eV) of O₂ on Ag₄⁺ in Fig. 3c, and the small amounts of reacted Ag₄⁺ in Fig. 1 can be rationalized using the kinetic barrier for the transition from the rhomboid to the pyramid of the silver frame or the presence of a tiny amount of the reactive pyramid isomer.

We also explored the structures of Ag_{*n*}O₂⁺ without restrictions that the two oxygen atoms connect together and silver clusters are maintained as one unit. The lowest-lying structures of this kind are shown in Fig. S12 (ESI†). The new structures determined for Ag_{*n*}O₂⁺ (*n* = 3–7) have higher energies than or energies very close to the ones shown in Fig. 3a. The energies of all even- and odd-sized Ag_{*n*}O₂⁺ (*n* = 8–12) in Fig. S12 (ESI†) are apparently lower than those of their structures in Fig. 3a, and the energy decreases from separate Ag_{*n*}⁺ and O₂ to these structures are much larger than the estimated threshold (~0.37 eV) resisting the entropy decrease. The completely different trends of these reaction energies in Fig. S12 (ESI†), and the reactivity shown in Fig. 1 imply that the dissociation of O₂ is less likely under present mild conditions. Using the reaction of Ag₁₀⁺ with O₂ as an example, we also considered reaction energies for possible etching reaction channels, which include those reported in previous experiments.^{35–39} The results shown in Fig. S13 (ESI†) indicate that all etching or dissociation channels are highly endothermic and cannot happen under present mild conditions.

The correlation between the H–L gap of Ag_{*n*}⁺ clusters and their reactivity to oxygen revealed in this study is in contrast to the reactivity of Ag_{*n*}[−] highly correlated with the VDE of clusters. The varying reactivity of anionic silver clusters with O₂ was generally attributed to two reasons:^{30–32,42,52} (1) the stable reaction products of odd-sized IB metal cluster anions (singlet) with O₂ (triplet) are in a singlet state, and there are always barriers related to the spin accommodation requirement in these reaction systems; (2) even-size IB metal cluster anions with lower electron detachment potentials are generally reactive, because their electronic properties facilitate the directed transfer of the unpaired electron from the metal moiety to bonded O₂. In previous studies of the reactions between Ag_{*n*}⁺ and O₂,^{35–39} observed Ag_{*n*}O₂⁺ complexes were considered to be

a combination of Ag_{*n*}²⁺ and O₂[−] through ionic bonding, where electron transfer similar to that in anionic silver or gold clusters was proposed. Following this assumption, the VDE value of clusters should be qualitatively consistent with their reactivity to O₂, as well as the amount of charge transfer from the cluster to the O₂ subunit. However, in the present study, there seems to be no apparent correlation between VDE values in Fig. 2b and NPA values in Fig. 3d or HOMO–LUMO gaps in Fig. 2d. The superatomic features in Fig. 4 cannot be associated with VDE values either. Instead, the exclusive dominance of H–L gaps of clusters on the reactivity was revealed, suggesting that the interaction in Ag_{*n*}O₂⁺ involves HOMO and LUMO of Ag_{*n*}⁺. We show orbital interactions of the silver moiety and O₂ molecule using two representative complexes, Ag₄O₂⁺ and Ag₆O₂⁺, in Fig. S14(a) and (b) (ESI†), respectively. In these two cases, the interaction between Ag_{*n*}⁺ and O₂ can be considered as follows: the single electron on the HOMO of Ag_{*n*}⁺ was excited to its LUMO, and the combination of this LUMO and one π_g of O₂ formed the bonding orbital in Ag_{*n*}O₂⁺. In a certain sense, this interaction has a covalent character.

3.4 Interplay between geometries and HOMO–LUMO gaps of Ag_{*n*}⁺ (*n* = 2–22)

An unexpected phenomenon in this study is the size-dependence of HOMO–LUMO gaps and therefore the reactivity of even sizes Ag_{*n*}⁺, which in many points are in marked contrast to the prediction of the free electron shell model.¹⁰ For example, even-sized Ag₄⁺ and Ag₁₀⁺ have one more electron than closed shells composed of 2 and 8 electrons, respectively, and therefore are expected to be like alkali metal atoms with small HOMO–LUMO gaps and high reactivity with O₂. This expectation is true for anionic Ag₂[−] (*n*_e = 2 + 1)^{29,53} and Ag₈[−] (*n*_e = 8 + 1),⁵⁴ both of which are quite reactive with O₂. However, experiments and calculations in this study showed that Ag₄⁺ and Ag₁₀⁺ have large HOMO–LUMO gaps and have low reactivity or are completely inert. The decrease of HOMO–LUMO gaps and the increase of the reactivity from Ag₄⁺ to Ag₆⁺ and from Ag₁₀⁺ to Ag₁₂⁺ are also unexpected. Ag₂₀⁺ and Ag₂₂⁺ are expected to have electron configurations like those of alkali metals, because they have one more electron than closed shells composed of 18 and 20 free electrons, respectively. However, Ag₂₀⁺ is found to have a quite large HOMO–LUMO gap and no reactivity with O₂. In Fig. 4, we plot energy levels of three pairs of even-sized species around unexpected turning points (Ag₄⁺ → Ag₆⁺; Ag₁₀⁺ → Ag₁₂⁺; Ag₂₀⁺ → Ag₂₂⁺). For Ag₄⁺, the lowest-lying rhomboid structure (in Fig. 2) and pyramid structure (an isomer 0.32 eV higher; shown in Fig. S2, ESI†) were considered, since chemisorbed O₂ on the rhomboid structure of Ag₄⁺ changed it to the pyramid one (shown in Fig. 3 and Fig. S11, ESI†). The energy levels and orbital shapes of pyramid Ag₄⁺ in Fig. 4a show that its three 1p orbitals are very close, and the electron configuration 1s²1p¹ leads to a very small HOMO–LUMO gap. The rhomboid Ag₄⁺ (in Fig. 4a) has only two 1p orbitals because of its planar potential field,¹¹ and the energies of these two orbitals are well separated because of the apparent difference between the two diagonals of the rhomboid. The 1p orbital along the longer diagonal is lower in



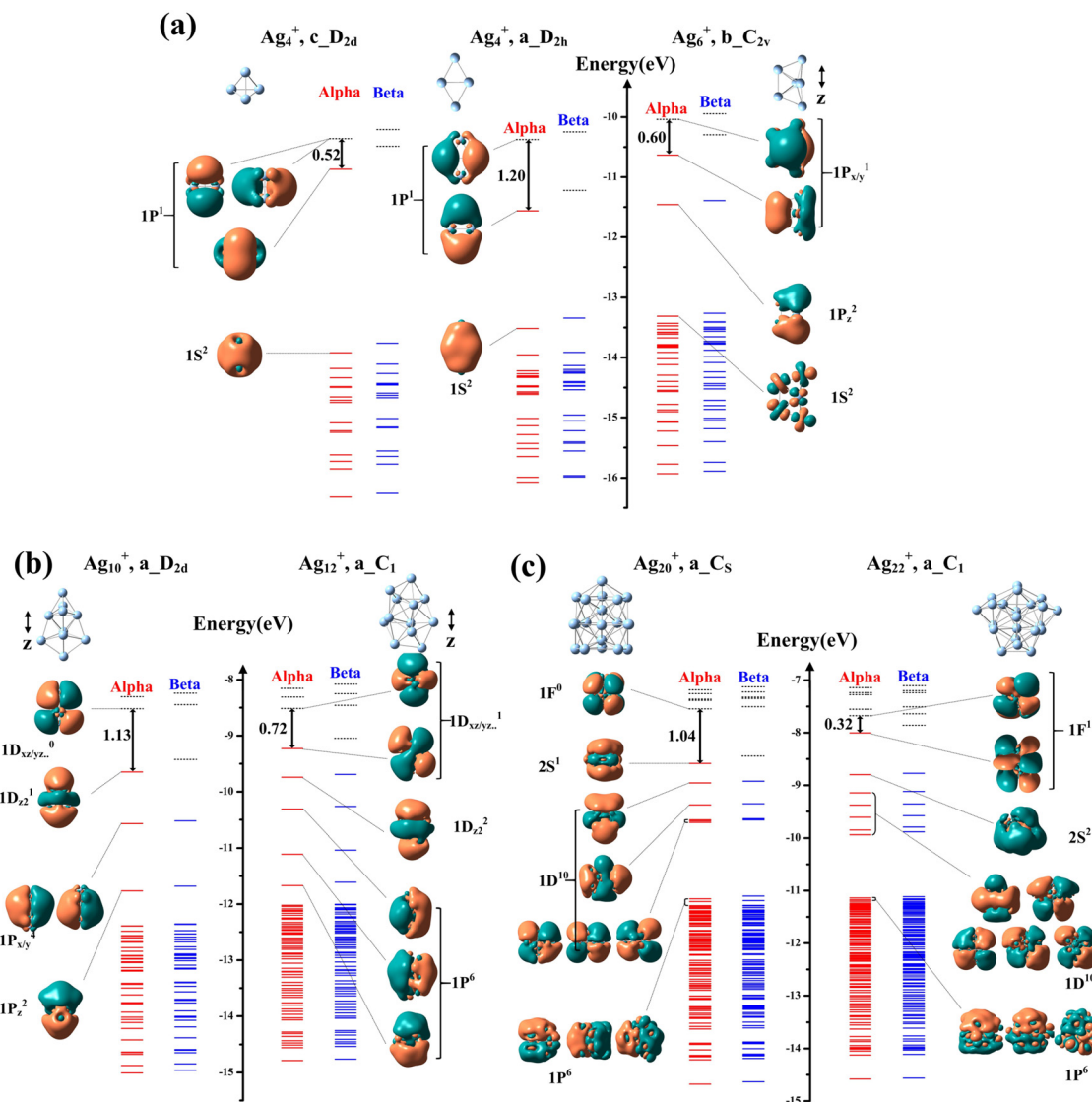


Fig. 4 Orbital levels in (a) Ag_4^+ (pyramid and rhomboid symmetries) and Ag_6^+ , (b) Ag_{10}^+ and Ag_{12}^+ , and (c) Ag_{20}^+ and Ag_{22}^+ . In each panel, red and the blue solid lines indicate occupied alpha and beta orbitals, respectively. Black dashed lines indicate unoccupied orbitals. The shapes of some orbitals around HOMOs and LUMOs are shown, and the isosurface value is ± 0.005 a.u. The naming of the orbitals was based on the free electron shell model. HOMO–LUMO gaps of alpha levels are indicated.

energy and the other along the shorter diagonal is higher. The geometries of Ag_6^+ (Fig. 4a), Ag_{10}^+ and Ag_{12}^+ (Fig. 4b) can roughly be viewed as a 3D structure extending longer in one direction (labeled as the z direction using arrows beside structures in Fig. 4) than other two orthogonal ones (x/y). Based on quantum mechanical figures, the orbital extending to the longer direction tends to be lower in energy than others of the same type. In Ag_6^+ (Fig. 4a), the electron configuration is $1s^2 1p_z^2 1p_{x/y}^1$, and the small energy difference between $1p_x$ and $1p_y$ leads to a small HOMO–LUMO gap. In Ag_{10}^+ (Fig. 4b), the electron configuration is $1s^2 1p_z^2 1p_{x/y}^4 1d_{z^2}^1 1d_{xy/yz/\dots}^1$, and the large separation between $1d_{z^2}$ and other $1d_{xy/yz/\dots}$ orbitals leads to a large HOMO–LUMO gap. In Ag_{12}^+ (Fig. 4b), the electron configuration is $1s^2 1p_z^2 1p_x^2 1p_y^2 1d_{z^2}^1 1d_{xy/yz/\dots}^1$ and the relatively small energy differences among $1d_{xy/yz/\dots}$ orbitals lead to a small HOMO–LUMO

gap. For Ag_{20}^+ and Ag_{22}^+ (Fig. 4c), the differences among their $x/y/z$ dimensions are insignificant, and the splitting of orbitals of the same type is relatively small. The electron configurations of Ag_{20}^+ and Ag_{22}^+ are $1s^2 1p^6 1d^{10} 2s^1$ and $1s^2 1p^6 1d^{10} 2s^2 1f^1$, respectively, and their HOMO–LUMO gaps correspond to the large gap between $2s$ and $1f$ shells and small gap between two $1f$ orbitals, respectively.

4. Conclusions

We presented a study on reactions of Ag_n^+ ($n = 2\text{--}22$) with O_2 by combining experimental measurements and theoretical calculations. Under mild conditions, the reactivity of Ag_n^+ showed a peculiar size dependence, in which only Ag_n^+ species



($n = 4, 6, 12, 16, 18,$ and 22) are reactive. The most likely structures of Ag_n^+ ($n = 2-22$) are determined using an improved genetic algorithm and a density functional theory method. The bonding strengths of O_2 on the determined structures of Ag_n^+ ($n = 2-22$) are consistent with experimental observations. Analyses of the electronic character of Ag_n^+ ($n = 2-22$) species indicated that their reactivity with O_2 is closely related to their HOMO–LUMO gaps. Theoretical results showed that decreased HOMO–LUMO gaps facilitated the formation of covalent interactions with O_2 under thermodynamic and kinetic considerations. The variation of HOMO–LUMO gaps and therefore the reactivity with O_2 was further attributed to a subtle interplay between electronic configurations and geometric structures of Ag_n^+ .

Conflicts of interest

There are no conflicts to declare.

Acknowledgements

This work was supported by the National Natural Science Foundation of China (Grant No. 22273065, 21673158) and Science & Technology Commission of Shanghai Municipality (14DZ2261100). Theoretical calculations were carried out on the computing resource in National Supercomputing Center in Shenzhen.

Notes and references

- S. M. Lang and T. M. Bernhardt, *Phys. Chem. Chem. Phys.*, 2012, **14**, 9255–9269.
- L.-M. Wang and L.-S. Wang, *Nanoscale*, 2012, **4**, 4038–4053.
- Z. X. Luo, A. W. Castleman and S. N. Khanna, *Chem. Rev.*, 2016, **116**, 14456–14492.
- P. Ferrari, J. Vanbuel, E. Janssens and P. Lievens, *Acc. Chem. Res.*, 2018, **51**, 3174–3182.
- J. L. Martins, J. Buttet and R. Car, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 1985, **31**, 1804–1816.
- M. Castro, C. Jamorski and D. R. Salahub, *Chem. Phys. Lett.*, 1997, **271**, 133–142.
- W. Satula, J. Dobaczewski and W. Nazarewicz, *Phys. Rev. Lett.*, 1998, **81**, 3599–3602.
- M. A. Tofanelli, K. Salorinne, T. W. Ni, S. Malola, B. Newell, B. Phillips, H. Hakkinen and C. J. Ackerson, *Chem. Sci.*, 2016, **7**, 1882–1890.
- Q. Y. Du, X. Wu, P. J. Wang, D. Wu, L. W. Sai, R. B. King, S. J. Park and J. J. Zhao, *J. Phys. Chem. C*, 2020, **124**, 7449–7457.
- W. A. Deheer, *Rev. Mod. Phys.*, 1993, **65**, 611–676.
- H. Hakkinen, *Adv. Phys.: X*, 2016, **1**, 467–491.
- T. Tsukamoto, T. Kambe, T. Imaoka and K. Yamamoto, *Nat. Rev. Chem.*, 2021, **5**, 338–347.
- R. M. Lambert, F. J. Williams, R. L. Copley and A. Palermo, *J. Mol. Catal. A: Chem.*, 2005, **228**, 27–33.
- Y. Lei, F. Mehmood, S. Lee, J. Greeley, B. Lee, S. Seifert, R. E. Winans, J. W. Elam, R. J. Meyer, P. C. Redfern, D. Teschner, R. Schlogl, M. J. Pellin, L. A. Curtiss and S. Vajda, *Science*, 2010, **328**, 224–228.
- H. I. Mahdi, N. N. Ramlee, D. H. D. Santos, D. A. Giannakoudakis, L. H. de Oliveira, R. Selvasembian, N. I. W. Azelee, A. Bazargan and L. Meili, *Mol. Catal.*, 2023, **537**, 112944.
- V. V. Dutov, G. V. Mamontov, V. I. Zaikovskii, L. F. Liotta and O. V. Vodyankina, *Appl. Catal., B*, 2018, **221**, 598–609.
- Z. P. Qu, M. Cheng, C. Shi and X. Bao, *J. Mol. Catal. A: Chem.*, 2005, **239**, 22–31.
- I. Katakuse, T. Ichihara, Y. Fujita, T. Matsuo, T. Sakurai and H. Matsuda, *Int. J. Mass Spectrom. Ion Processes*, 1985, **67**, 229–236.
- G. Alameddin, J. Hunter, D. Cameron and M. M. Kappes, *Chem. Phys. Lett.*, 1992, **192**, 122–128.
- K. J. Taylor, C. L. Pettiette-Hall, O. Cheshnovsky and R. E. Smalley, *J. Chem. Phys.*, 1992, **96**, 3319.
- H. Handschuh, C. Y. Cha, P. S. Bechthold, G. Ganteför and W. Eberhardt, *J. Chem. Phys.*, 1995, **102**, 6406–6422.
- K. Majer and B. von Issendorff, *Phys. Chem. Chem. Phys.*, 2012, **14**, 9371–9376.
- P. Weis, T. Bierweiler, S. Gilb and M. M. Kappes, *Chem. Phys. Lett.*, 2002, **355**, 355–364.
- H. Hakkinen, M. Moseler, O. Kostko, N. Morgner, M. A. Hoffmann and B. von Issendorff, *Phys. Rev. Lett.*, 2004, **93**, 093401.
- X. P. Xing, R. M. Danell, I. L. Garzon, K. Michaelian, M. N. Blom, M. M. Burns and J. H. Parks, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2005, **72**, 081405.
- D. Schooss, M. N. Blom, J. H. Parks, B. von Issendorff, H. Haberland and M. M. Kappes, *Nano Lett.*, 2005, **5**, 1972–1977.
- M. N. Blom, D. Schooss, J. Stairs and M. M. Kappes, *J. Chem. Phys.*, 2006, **124**, 244308.
- T. H. Lee and K. M. Ervin, *J. Phys. Chem.*, 1994, **98**, 10023–10031.
- J. Hagen, L. D. Socaciu, J. LeRoux, D. Popolan, T. M. Bernhardt, L. Woste, R. Mitric, H. Noack and V. Bonacic-Koutecky, *J. Am. Chem. Soc.*, 2004, **126**, 3442–3443.
- Z. X. Luo, G. U. Gamboa, J. C. Smith, A. C. Reber, J. U. Reveles, S. N. Khanna and A. W. Castleman, *J. Am. Chem. Soc.*, 2012, **134**, 18973–18978.
- J. Ma, X. Cao, X. Xing, X. Wang and J. H. Parks, *Phys. Chem. Chem. Phys.*, 2016, **18**, 743–748.
- B. Q. Yin, Q. Y. Du, L. J. Geng, H. Y. Zhang, Z. X. Luo, S. Zhou and J. J. Zhao, *CCS Chem.*, 2021, **3**, 219–229.
- W. Liu, L. L. Huang, L. Meng, J. Hu and X. P. Xing, *Phys. Chem. Chem. Phys.*, 2023, **25**, 14303–14310.
- Q. Y. Du, L. L. Huang, J. Q. Fu, Y. J. Cao, X. P. Xing and J. J. Zhao, *J. Chem. Phys.*, 2023, **158**, 014306.
- M. Schmidt, P. Cahuzac, C. Brechignac and H. P. Cheng, *J. Chem. Phys.*, 2003, **118**, 10956–10962.
- M. Schmidt, A. Masson and C. Brechignac, *Phys. Rev. Lett.*, 2003, **91**, 243401.
- Y.-N. Wu, M. Schmidt, J. Leygnier, H.-P. Cheng, A. Masson and C. Brechignac, *J. Chem. Phys.*, 2012, **136**, 024314.



- 38 M. Schmidt, A. Masson, H.-P. Cheng and C. Brechignac, *ChemPhysChem*, 2015, **16**, 855–865.
- 39 M. Schmidt and C. Brechignac, *C. R. Phys.*, 2016, **17**, 481–484.
- 40 M. Arakawa, N. Hayashi, K. Minamikawa, T. Nishizato and A. Terasaki, *J. Phys. Chem. A*, 2022, **126**, 6920–6926.
- 41 J. van der Tol, D. Jia, Y. Li, V. Chernyy, J. M. Bakker, N. Minh Tho, P. Lievens and E. Janssens, *Phys. Chem. Chem. Phys.*, 2017, **19**, 19360–19368.
- 42 T. Wang, J. Ma, B. Yin and X. Xing, *J. Phys. Chem. A*, 2018, **122**, 3346–3352.
- 43 M. J. Frisch, G. W. Trucks, H. B. Schlegel, G. E. Scuseria and M. A. Robb, *et al.*, *Gaussian 09 Rev. C.01*, Wallingford, CT, 2010.
- 44 B. P. Pritchard, D. Altarawy, B. Didier, T. D. Gibson and T. L. Windus, *J. Chem. Inf. Model.*, 2019, **59**, 4814–4820.
- 45 D. Andrae, U. Haussermann, M. Dolg, H. Stoll and H. Preuss, *Theor. Chim. Acta*, 1990, **77**, 123–141.
- 46 F. Weigend and R. Ahlrichs, *Phys. Chem. Chem. Phys.*, 2005, **7**, 3297–3305.
- 47 H. L. Zhang and D. X. Tian, *Comput. Mater. Sci.*, 2008, **42**, 462–469.
- 48 Y. Y. Jin, Y. H. Tian, X. Y. Kuang, C. Z. Zhang, C. Lu, J. J. Wang, J. Lv, L. P. Ding and M. Ju, *J. Phys. Chem. A*, 2015, **119**, 6738–6745.
- 49 M. L. McKee and A. Samokhvalov, *J. Phys. Chem. A*, 2017, **121**, 5018–5028.
- 50 S.-Y. Yan, W. Zhang, Z.-X. Zhao, W.-C. Lu and H.-X. Zhang, *Theor. Chem. Acc.*, 2012, **131**, 1200.
- 51 L. Huang, W. Liu, J. Hu and X. Xing, *J. Phys. Chem. A*, 2021, **125**, 9995–10005.
- 52 B. E. Salisbury, W. T. Wallace and R. L. Whetten, *Chem. Phys.*, 2000, **262**, 131–141.
- 53 T. M. Bernhardt, J. Hagen, S. M. Lang, D. M. Popolan, L. D. Socaciu-Siebert and L. Woste, *J. Phys. Chem. A*, 2009, **113**, 2724–2733.
- 54 Z. Luo, C. Berkdemir, J. C. Smith and A. W. Castleman, *Chem. Phys. Lett.*, 2013, **582**, 24–30.

