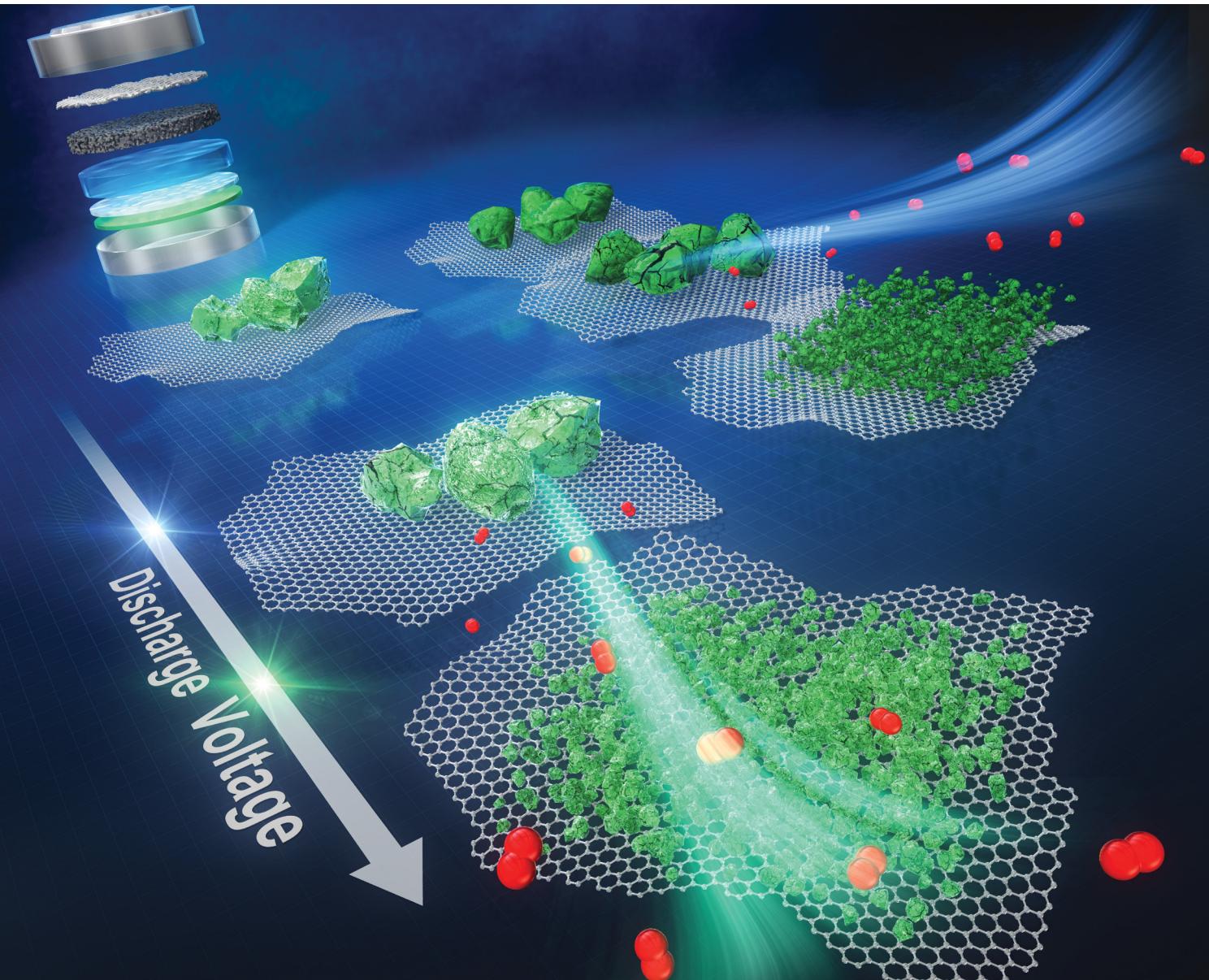


# PCCP

Physical Chemistry Chemical Physics

[rsc.li/pccp](http://rsc.li/pccp)

**25**  
YEARS  
ANNIVERSARY



ISSN 1463-9076



Cite this: *Phys. Chem. Chem. Phys.*, 2024, 26, 13655

# Nature of $\text{Li}_2\text{O}_2$ and its relationship to the mechanisms of discharge/charge reactions of lithium–oxygen batteries

Yanan Gao, <sup>ac</sup> Hitoshi Asahina, <sup>ab</sup> Shoichi Matsuda, <sup>ab</sup> Hidenori Noguchi <sup>ac</sup> and Kohei Uosaki <sup>\*ab</sup>

Lithium–air batteries (LABs) are considered one of the most promising energy storage devices because of their large theoretical energy density. However, low cyclability caused by battery degradation prevents its practical use. Thus, to realize practical LABs, it is essential to improve cyclability significantly by understanding how the degradation processes proceed. Here, we used online mass spectrometry for real-time monitoring of gaseous products generated during charging of lithium–oxygen batteries (LOBs), which was operated with pure oxygen not air, with 1 M lithium bis(trifluoromethanesulfonyl)-imide (LiTFSI) tetraethylene glycol dimethyl ether (TEGME) electrolyte solution. Linear voltage sweep (LVS) and voltage step modes were employed for charge instead of constant current charge so that the energetics of the product formation during the charge process can be understood more quantitatively. The presence of two distinctly different types of  $\text{Li}_2\text{O}_2$ , one being decomposed in a wide range of relatively low cell voltages (2.8–4.16 V) (l- $\text{Li}_2\text{O}_2$ ) and the other being decomposed at higher cell voltages than ca. 4.16 V (h- $\text{Li}_2\text{O}_2$ ), was confirmed by both LVS and step experiments.  $\text{H}_2\text{O}$  generation started when the  $\text{O}_2$  generation rate reached a first maximum and  $\text{CO}_2$  generation took place accompanied by the decomposition of h- $\text{Li}_2\text{O}_2$ . Based on the above results and the effects of discharge time and the use of isotope oxygen during discharge on product distribution during charge, the generation mechanism of  $\text{O}_2$ ,  $\text{H}_2\text{O}$ , and  $\text{CO}_2$  during charging is discussed in relation to the reactions during discharge.

Received 30th January 2024,  
Accepted 18th March 2024

DOI: 10.1039/d4cp00428k

rsc.li/pccp

## 1. Introduction

Rechargeable batteries are considered to play key roles in a future sustainable, carbon neutral society based on renewable energy as energy sources for fossil fuel-free transportation and as energy storage devices for stabilizing the temporal/spatial variation of the power generation by solar cells, windmills, and other renewable/natural energy sources.<sup>1–5</sup> There are, however, many issues to be considered for the widespread use of rechargeable batteries such as energy/power densities, cycle/calendar life, cost, and resource constraints.<sup>5–10</sup>

Lithium–air batteries (LABs) have attracted significant interest over the past decades because of their very high theoretical energy density and transition metal-free positive electrode (cathode during discharge).<sup>11–22</sup> Oxygen from air and Li metal

are the active materials for the cathode and anode, respectively, for LABs with the cell reaction of  $2\text{Li} + \text{O}_2 \rightleftharpoons \text{Li}_2\text{O}_2$ . Usually, porous carbon is used as a positive electrode material to accommodate  $\text{Li}_2\text{O}_2$  generated during discharge.<sup>23–28</sup> There are many critical problems to be solved before LABs can be used as a practical device. Many components of air such as water,  $\text{CO}_2$ , and even  $\text{N}_2$ , are harmful to LAB operation<sup>29–32</sup> and, therefore, most fundamental studies have been carried out for lithium oxygen batteries (LOBs), which use pure dry  $\text{O}_2$  instead of air.<sup>29,33,34</sup> Even LOBs have serious problems such as low cyclability and stability mainly due to the degradation of the positive electrode (carbon) and electrolyte solution, and the degradation/dendrite formation of the negative Li metal electrode.<sup>35–41</sup> To improve the cyclability/stability, it is essential to clarify the mechanism of LOB degradation.<sup>42,43</sup> While the degradation/dendrite formation of Li metal electrode is a common issue of all next generation batteries using Li metal as a negative electrode (anode during discharge), degradation of the positive electrode and electrolyte is more serious in LOBs than in other next generation batteries because  $\text{Li}_2\text{O}_2$ , the product at the cathode during discharge of the LOB, is an insulator, which requires a high overpotential to be

<sup>a</sup> Center for Green Research on Energy and Environmental Materials, National Institute for Materials Science (NIMS), Namiki 1-1, Tsukuba 305-0044, Japan.  
E-mail: uosaki.kohei@nims.go.jp

<sup>b</sup> SoftBank-NIMS Advanced Technologies Development Center, NIMS, 1-1 Namiki, Tsukuba, Ibaraki 305-0044, Japan

<sup>c</sup> Graduate School of Chemical Sciences and Engineering, Hokkaido University, Sapporo 060-0810, Japan



decomposed during charge, and active oxygen species such as  $O_2^-$  and  $^1O_2$  generated during discharge and charge attack and degrade the electrolyte and carbon positive electrode.<sup>35-39</sup> Many efforts have been made to investigate the degradation mechanism at the positive electrode using various techniques such as surface enhanced Raman scattering,<sup>12,36,44</sup> electrochemical quartz crystal microbalance,<sup>45</sup> X-ray diffraction (XRD),<sup>14,21,25,36,42,46,48,49</sup> X-ray photoelectron spectroscopy (XPS),<sup>47,48</sup> and mass spectrometry.<sup>21,36,40,42,49-55</sup>

Reaction sites for the formation of  $Li_2O_2$  have been discussed for a long time but still no agreement has been reached. Recently we reported that a very long ( $\sim 80\ \mu m$ )  $Li_2O_2$  nanowire can be grown at a gold electrode covered with single layer graphene (SLG) and the reaction site is the defect sites of SLG on the Au electrode/ $Li_2O_2$  interface based on Raman analysis with isotopic  $^{18}O_2$ .<sup>56</sup> On the other hand, Nakanishi and his colleagues suggested that  $Li_2O_2$  grew at the electrolyte/ $Li_2O_2$  interface of a surface modified carbon paper cathode during discharge based on the results of nano-SIMS analysis and that there exist two types of  $Li_2O_2$ ; one is formed close to the electrode in the earlier stage of discharge and is more difficult to be oxidized and the other is formed in the latter stage of discharge covering the  $Li_2O_2$  which is formed earlier and is more easily oxidized.<sup>57</sup> More recently, Tan *et al.* suggested that  $Li_2O_2$  deposition takes place both at the cathode/ $Li_2O_2$  interface and at the electrolyte/ $Li_2O_2$  interface after the initial deposition of insulating  $Li_2O_2$  during discharge.<sup>58,59</sup>

The origin of  $CO_2$  is also a very important issue for understanding the degradation mechanism.<sup>60-66</sup> Many papers have mentioned that  $CO_2$  is originated from carbonate and carboxylate species such as  $Li_2CO_3$ ,  $HCO_2Li$ , and  $CH_3CO_2Li$ , which are formed by the attack of active oxygen species on electrolyte and/or carbon during discharge/charge processes.<sup>64-66</sup> McCoskey *et al.* suggested that about 50% of  $CO_2$  are originated from carbonate, which is generated by the oxidation of carbon positive electrode, based on the results obtained using a carbon positive electrode made of  $^{13}C$ .<sup>63</sup>

In the present work, we used online mass spectrometry for real time monitoring of generated gaseous products during charging of LOBs with 1 M lithium bis(trifluoromethane-sulfonyl)imide (LiTFSI) tetraethylene glycol dimethyl ether (TEGDME) electrolyte solution. Discharge of LOBs was carried out for various time periods and under the flow of two isotope  $O_2$  gas,  $^{16}O_2$  and  $^{18}O_2$ . The  $O_2$  generation during charging showed the presence of two distinctly different types of  $Li_2O_2$ , one being decomposed in a wide range of relatively low cell voltages (2.8 – *ca.* 4.1 V) (l- $Li_2O_2$ ) and the other being decomposed at cell voltages higher than *ca.* 4.1 V (h- $Li_2O_2$ ) during charging.  $H_2O$  and  $CO_2$  generation took place at relatively low voltage accompanied by the decomposition of l- $Li_2O_2$  and at higher cell voltage with the decomposition of h- $Li_2O_2$ , respectively. Based on the above results and the effects of discharge time and the use of isotope oxygen during discharge on product distribution during charge, the generation mechanism of  $O_2$ ,  $H_2O$ , and  $CO_2$  during charge is discussed in relation to the reactions during discharge.

## 2. Experimental section

### 2.1. Materials

A CNT-based porous carbon sheet (KJCN, thickness 50  $\mu m$ ) donated by KJ Specialty Paper was used as a positive electrode as it has more uniform and simpler pore structures than Ketjenblack and carbon nanotube-based positive electrodes which we usually use for more practical cells.<sup>54,55</sup> The KJCN sheet was cut into discs with a diameter of 16 mm, baked in a tube furnace for 3 h at 900 °C under an Ar atmosphere, then transferred to a dry room for cell assembly. Disk-shaped lithium foil (thickness 200  $\mu m$ , diameter 16 mm) obtained from Honjo Metal was used as a negative electrode. Battery grade TEGDME and LiTFSI were obtained from Kishida Chemical and used as received. 1 M LiTFSI TEGDME prepared in a super dry room was used as an electrolyte solution. The water content in electrolyte solutions was around 40 ppm as measured by using a Karl Fischer Moisture Meter (Model: CA-21; Mitsubishi Chemical Analytech). A polyethylene (PE) membrane separator (thickness 20  $\mu m$ ) was obtained from W-scope and was cut into a disc with a diameter of 19.5 mm and then dried in a vacuum oven for 10 h at 40 °C before use. A stainless steel mesh (thickness 200  $\mu m$ , diameter 16.7 mm, aperture ratio 73%) was obtained from Hohsen. Isotope  $^{18}O_2$  gas (purity  $\geq 98$  atom%) was purchased from Taiyo Nippon Sanso.

### 2.2. Cell assembly and measurement system

Fig. 1 shows the experimental set up for the gas analysis during discharge and charge.<sup>50</sup> A lithium–oxygen cell was assembled in a stainless steel container (inner diameter: 45 mm, inner depth: 15 mm) with a lid equipped with a gas inlet and outlet by stacking Li metal foil, a polyethylene membrane separator, on top of which 15  $\mu L$  of the 1 M LiTFSI TEGDME solution was placed as uniformly as possible, a KJCN sheet, a stainless steel mesh as a current collector for the positive electrode, and a conductive spring coil, which provided the cell assembly with a pressure of 35 kPa as shown in Fig. 1. The apparent electrode area was 2  $cm^2$ . The assembly was carried out in a dry-booth (dew point less than –90 °C) in a dry room (dew point less than –60 °C).

The cell was connected to an inlet gas line with a three-way Swagelok valve, V1, to select  $O_2$  and He gas during discharge and charge, respectively, electrochemical controllers (a charge/discharge system: HJ1020mSD8, Hokuto Denko and an electrochemical measurement system: HZ-7000, Hokuto Denko), and

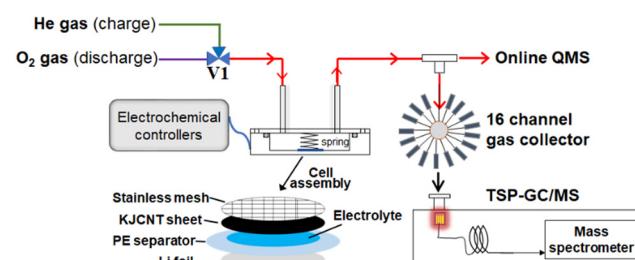


Fig. 1 Schematic diagram of experimental set-up and cell assembly.



an outlet mass analysis line, which had two branch lines; one to a mass spectrometer (Canon Anelva Quadrupole Mass Spectrometer M-401GA-DM) through a capillary tube (internal diameter: 0.05 mm, length: 4 m) for online real time analysis and the other to a gas collector for post TSP-GC/MS analysis of organic products, the results of which are not discussed in this study.

### 2.3. Analysis methods

After the cell was connected to the measurement system, the cell was purged by flowing  $O_2$  at  $2 \text{ cc min}^{-1}$  for 2 h and then constant current discharge was carried out under  $O_2$  flow at  $2 \text{ cc min}^{-1}$  by using the discharge/charge system with a current density of  $0.4 \text{ mA}$  ( $0.2 \text{ mA cm}^{-2}$ ) for 10 h or until reaching the cut-off voltage of 2 V. After the discharge was completed, the cell was set to open circuit and the inlet gas was switched from  $O_2$  to He.  $O_2$  in the cell was totally exchanged by He by flowing He by  $20 \text{ cc min}^{-1}$  for 90 min. The flow rate of He gas was then reduced to  $5 \text{ cc min}^{-1}$  and online mass spectrometry for mass numbers ( $m/z$ ) between 12 and 90 was started. After 30 min He flow at OCP to stabilize the background mass level, the charge was performed under constant current ( $0.4 \text{ mA}$  ( $0.2 \text{ mA cm}^{-2}$ )) for 10 h or until reaching the cut-off voltage of 4.8 V using the discharge/charge system, linear voltage sweep (LVS:  $0.05 \text{ mV s}^{-1}$  up to 4.8 V) using the potentiostat/galvanostat, or voltage step using the potentiostat/galvanostat. When the potentiostat/galvanostat was used, the carbon positive electrode was connected to the working electrode terminal and Li metal negative electrode was connected to both the counter and reference electrode terminals of the potentiostat/galvanostat so that the voltage applied to the cell was equivalent to the potential of the carbon electrode with respect to that of the Li electrode.

## 3. Results and discussions

### 3.1. Constant current discharge/charge curves and mass signals during charge

Fig. 2 shows the typical 1st discharge and charge curves (bottom panel) and ion currents for  $m/z = 18$  ( $H_2O$ ),  $32$  ( $O_2$ ), and  $44$  ( $CO_2$ ) as a function of charge of the present cell configuration obtained with a constant current of  $0.2 \text{ mA cm}^{-2}$ .

As already reported by many groups including ours for various carbon positive electrodes, while the discharge voltage was relatively constant with a small overpotential up to  $2 \text{ mA h cm}^{-2}$  of discharge, the voltage increased significantly with several inflection points during charge, showing several processes with different redox potentials were involved in the charge process. The charge dependencies of the signals of  $m/z = 18$  ( $H_2O$ ),  $32$  ( $O_2$ ), and  $44$  ( $CO_2$ ) were essentially the same as those reported in our previous papers,<sup>50</sup> although the previous result was for a LOB with  $NO_3^-/Br^-$  dual mediators and a Ketjenblack positive electrode. The charge dependence of the mass signal for  $O_2$  ( $m/z = 32$ ) is not simple. The signal of  $m/z = 32$  increased sharply as the charge started, reached a peak,

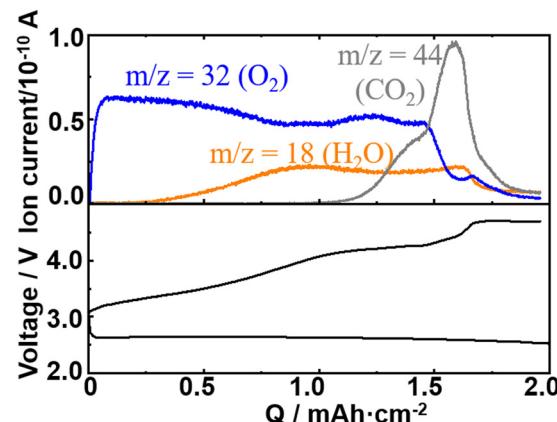


Fig. 2 Typical discharge/charge curves (bottom panel) and ion currents of the Q-mass at  $m/z = 18$  ( $H_2O$ ),  $32$  ( $O_2$ ), and  $44$  ( $CO_2$ ) during charge (top panel) of a Li/TEGDME-1M LiTFSI/KJCNt cell (electrode area:  $2 \text{ cm}^2$ ) obtained under a constant current of  $0.4 \text{ mA}$  ( $0.2 \text{ mA cm}^{-2}$ ).

and declined but then increased and decreased again despite the constant current flow, showing that the current efficiency for the oxidation of  $Li_2O_2$ , *i.e.*, the formation of  $O_2$ , varied during the charge in a complex manner. It seems the increase and decrease of the signal of  $m/z = 32$  ( $O_2$ ) were related to the variations of signals of  $m/z = 18$  ( $H_2O$ ) and  $44$  ( $CO_2$ ), suggesting that the variation of the current efficiency for the oxidation of  $Li_2O_2$  was affected by the generations of  $H_2O$  and  $CO_2$ .

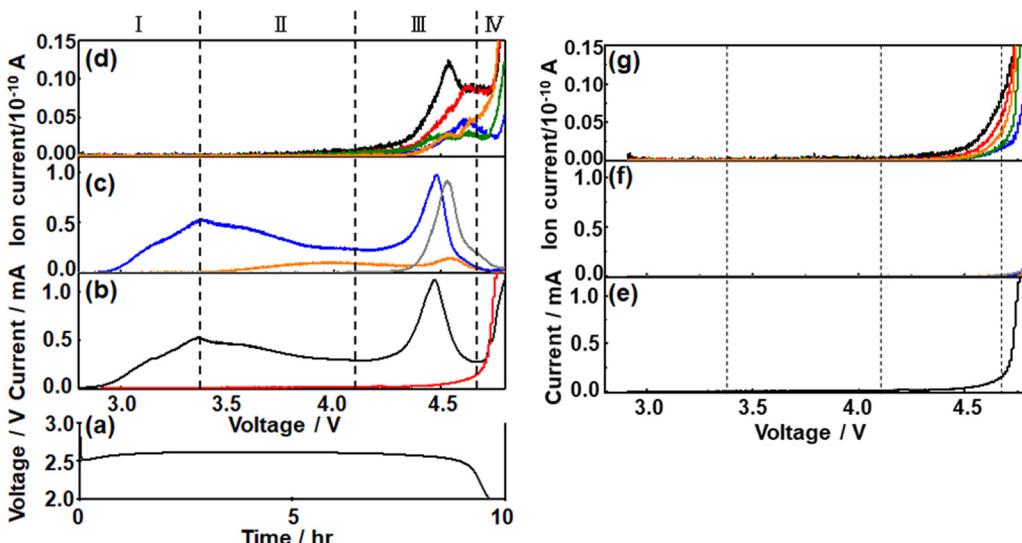
### 3.2. Behaviours of current and mass signals during LVS charge from OCP to 4.8 V after constant current discharge

To understand the charge process more quantitatively, the charge was performed under LVS instead of constant current.<sup>49,60,67-69</sup>

After constant current discharge ( $0.2 \text{ mA cm}^{-2}$ ) for 9.6 h (cut-off voltage of 2 V as shown in Fig. 3(a)), the cell voltage was scanned from OCP, *i.e.*, 2.8 V, to 4.8 V with a scan rate of  $0.05 \text{ mV s}^{-1}$  and current and online mass signals were recorded simultaneously. Fig. 3(b) shows the current (black line) as a function of the voltage. Current increased immediately as the voltage started to be scanned from OCP and reached a maximum at 3.36 V. It decreased gradually after the 1st maximum, reached the 1st minimum at 4.15 V, increased again to reach the 2nd maximum at 4.46 V, decreased to reach the 2nd minimum at 4.66 V, and then increased again. The second current peak declined sharply with rather symmetric shape. The current–voltage relation obtained after discharge can be divided into four regions; Region I: up to the 1st maximum (~3.36 V), Region II: from the 1st maximum to the 1st minimum (3.36–4.15 V), Region III: from the 1st minimum to the 2nd minimum (4.15–4.66 V), and Region IV: from the 2nd minimum (4.66–4.8 V, *i.e.*, final current increase).

The current response of the same cell configuration but without discharge under LVS with the scan rate of  $0.05 \text{ mV s}^{-1}$  is also shown in Fig. 3(b) (red line) and Fig. 3(e). In this case, the current was very small compared with that observed in the discharged cell (black line in Fig. 3(b)) up to 4.15 V (in Regions I and II of the discharged cell), started to increase from around





**Fig. 3** (a) Discharge curve of a Li/TEGDME-1M LiTFSI/KJCNT cell (electrode area:  $2\text{ cm}^2$ ) obtained under a constant current of  $0.4\text{ mA}$  ( $0.2\text{ mA cm}^{-2}$ ). (b and e) Current, (c and f) signal intensities of  $m/z = 18$  ( $\text{H}_2\text{O}^+$ ): orange,  $m/z = 32$  ( $\text{O}_2^+$ ): blue, and  $m/z = 44$  ( $\text{CO}_2^+$ ): gray (middle panel), and (d and g) intensities of mass signals of fragments related to organic molecules derived from TEGDME, *i.e.*,  $m/z = 28$  ( $\text{CO}^+$ ): black,  $m/z = 29$  ( $\text{COH}^+$ ): red,  $m/z = 30$  ( $\text{HCOH}^+$ ): blue,  $m/z = 31$  ( $\text{CH}_3\text{O}^+$ ,  $\text{CH}_2\text{OH}^+$ ): green, and  $m/z = 45$  ( $\text{CH}_3\text{OCH}_2^+$ ,  $\text{COOH}^+$ ): orange, as a function of cell voltage during the potential sweep with  $0.05\text{ mV s}^{-1}$  after (b-d) and without discharge (e-g). The current response without discharge (e) is also shown in (b) (red line) for a comparison. Note: the scales for the y-axis of (c and f) are 8 times larger than those of (d and g).

4.15 V, first gradually and then significantly in the potential region corresponding to Region III and Region IV, respectively, of the discharged cell. The current observed at the cell without discharge should be due to the direct anodic decomposition of the components of the electrolyte solution, TEGDME and LiTFSI. It became larger than that in the discharged cell when the voltage became more than *ca.* 4.7 V.

These results suggest that the current in the discharged cell in Region I-III was mostly due to the anodic oxidation of discharged products, mainly  $\text{Li}_2\text{O}_2$ , with a slight contribution of the direct electrochemical decomposition of the electrolyte solutions and that in Region IV was almost due to the direct electrochemical anodic decomposition of the electrolyte solutions.

The signal of  $m/z = 32$  ( $\text{O}_2$ ) behaved similarly to the current and two maxima were observed. It increased immediately as did the current when the voltage scan was started and reached a maximum at 3.36 V (Region I). The signals of  $m/z = 18$  ( $\text{H}_2\text{O}$ ) and  $m/z = 44$  ( $\text{CO}_2$ ) were not significant in Region I. The major reaction in Region I is  $\text{Li}_2\text{O}_2 \rightarrow 2\text{Li}^+ + \text{O}_2 + 2\text{e}^-$ .

In Region II, the  $\text{O}_2$  mass signal ( $m/z = 32$ ) decreased gradually after the 1st maximum as did the current but declined more rapidly than the current and reached a minimum at 4.15 V. A steeper decline of the  $\text{O}_2$  mass signal than the current means a decrease of the current efficiency for  $\text{O}_2$  generation by  $\text{Li}_2\text{O}_2 \rightarrow 2\text{Li}^+ + \text{O}_2 + 2\text{e}^-$  in this region. The  $\text{H}_2\text{O}$  mass signal ( $m/z = 18$ ) started to increase just after the 1st maxima of current and  $\text{O}_2$  signal where the current efficiency for  $\text{O}_2$  generation started to decrease, suggesting the  $\text{H}_2\text{O}$  formation is responsible for the decrease of the current efficiency for oxygen generation, *i.e.*,  $\text{Li}_2\text{O}_2$  oxidation. The plausible source of  $\text{H}_2\text{O}$  generation is  $\text{LiOH}$  with

the reaction of  $\text{LiOH} \rightarrow \text{Li}^+ + 1/2\text{H}_2\text{O} + 1/4\text{O}_2 + \text{e}^-$  (*ca.* 3.4 V).<sup>66</sup>  $\text{LiOH}$  can be generated during discharge.<sup>50</sup> The mass signal due to  $\text{CO}_2$  ( $m/z = 44$ ) was not significant in Region II.

In Region III, the  $\text{O}_2$  signal increased again after the minimum at 4.15 V and reached a maximum at 4.48 V, which is 20 mV more than the position of the 2nd current peak, and decreased as current but more rapidly. This suggests the presence of another type of  $\text{Li}_2\text{O}_2$ , which is more difficult to oxidize than  $\text{Li}_2\text{O}_2$  oxidized in Region I and II. The  $\text{H}_2\text{O}$  signal declined very slightly with voltage and started to increase at *ca.* 4.4 V to reach a maximum at *ca.* 4.55 V, which is more than the peak positions of the current (4.46 V) and  $\text{O}_2$  ( $m/z = 32$ ) signal (4.48 V). The  $\text{CO}_2$  signal ( $m/z = 44$ ) started to increase at 4.15 V, where the current and the  $\text{O}_2$  signal started to increase again, and reached a maximum at *ca.* 4.53 V, which was more than the peak positions of the current and  $\text{O}_2$  signal but was slightly less than that of the  $\text{H}_2\text{O}$  peak. The plausible source and reaction of  $\text{CO}_2$  generation are  $\text{Li}_2\text{CO}_3$  and  $\text{Li}_2\text{CO}_3 \rightarrow 2\text{Li}^+ + \text{CO}_2 + 1/2\text{O}_2 + 2\text{e}^-$  (*ca.* 3.8 V), respectively.<sup>66</sup>

Thus,  $\text{Li}_2\text{O}_2$  was decomposed in Region I, II, and III and the presence of two distinct  $\text{O}_2$  signal peaks (3.36 V and 4.48 V) shows that there existed at least two types of  $\text{Li}_2\text{O}_2$ , the one oxidized at lower voltages (in Regions I and II and some in Region III:  $\text{l-Li}_2\text{O}_2$ ) and the other oxidized at higher voltages (Region III:  $\text{h-Li}_2\text{O}_2$ ) as suggested by several previous reports.<sup>70-73</sup> While  $\text{l-Li}_2\text{O}_2$  was oxidized in a wide range of voltages with a very broad peak (Region I and II),  $\text{h-Li}_2\text{O}_2$  was oxidized in a relatively narrow range of voltages with a rather sharp symmetric peak with full width at half maximum of *ca.* 0.14 V. These results suggest that the natures and oxidation mechanisms of  $\text{l-Li}_2\text{O}_2$  and  $\text{h-Li}_2\text{O}_2$  were quite different.



While l-Li<sub>2</sub>O<sub>2</sub> was energetically distributed very widely with the required oxidation potentials from 2.8 V to 4.4 V or more, the energetic states of h-Li<sub>2</sub>O<sub>2</sub> were rather sharp with an oxidation potential of around 4.5 V.

In Region IV, the mass signals of O<sub>2</sub>, H<sub>2</sub>O, and CO<sub>2</sub> declined monotonously, although the current increased again rapidly at potentials more positive than 4.66 V. Instead, signals related to organic molecules derived from TEGDME such as *m/z* = 29, 31, 45, and 60 became significant in this region as shown in Fig. 3(d). These results agreed with the above suggestion that the current of the discharged cell in Region IV was mainly due to the direct electrochemical decomposition of the electrolyte solutions.

The contribution of the direct electrochemical decomposition of the electrolyte solutions to the mass signals – voltage relationships of the discharged cell can be evaluated by comparing with those of the same cell configuration but without discharge obtained under LVS with the scan rate of 0.05 mV s<sup>-1</sup> shown in Fig. 3(f) and (g). At the cell without discharge, the current started to increase from around 4.15 V and increased gradually and significantly in the potential region corresponding to Region III and Region IV, respectively, of the discharged cell as already mentioned (Fig. 3(b) and (e)). Mass signals related to organic molecules derived from TEGDME such as *m/z* = 29, 31, 45, and 60 (Fig. 3(g)) increased accordingly but those of *m/z* = 18, 32, and 44 (Fig. 3(f)) were almost negligible compared with that of the discharged cell in Regions I–III. These results confirm that the products detected in the discharged cell in Region III were mainly due to the anodic oxidation of discharged products, mainly Li<sub>2</sub>O<sub>2</sub>, with a slight contribution of the direct electrochemical anodic decomposition of the components of the electrolyte solution, TEGDME and LiTFSI, and those in Region IV were almost due to the direct electrochemical anodic decomposition of the electrolyte solution, TEGDME and LiTFSI, as already suggested based on current–voltage relationships.

#### Contribution of various oxidation reactions on current during LVS charge of the discharged cell

As mentioned above, after the 1st peak, the O<sub>2</sub> mass signal declined steeper than the current with potential. This can be seen more clearly in Fig. 4 where partial current due to O<sub>2</sub> generation, which was obtained by fitting the voltage dependence of O<sub>2</sub> mass signal normalized at the 1st peak, was plotted (blue line) along with current (black line) as a function of voltage. Although there existed a slight time delay of partial current corresponding to O<sub>2</sub> generation relative to current because of the delay between generation and mass detection of O<sub>2</sub>, total current and partial current due to O<sub>2</sub> generation are matched reasonably well up to the 1st peak.

After the 1st peak, the partial current (blue line) due to O<sub>2</sub> generation became less than the total current (black line). Since the H<sub>2</sub>O signal increased in this region as mentioned above, a plausible reason for this discrepancy between the measured current and partial current due to O<sub>2</sub> generation is the partial current due to H<sub>2</sub>O generation, which was evaluated by utilizing

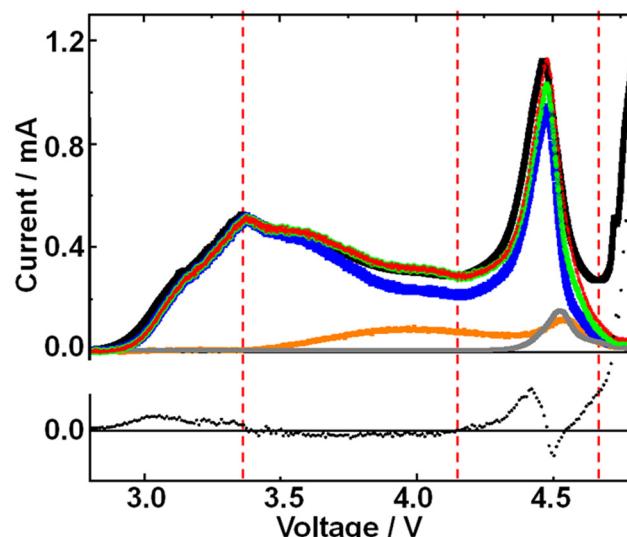


Fig. 4 Top panel: Measured current (black), partial currents due to O<sub>2</sub> generation (blue), H<sub>2</sub>O generation (orange), and CO<sub>2</sub> generation (gray), and the sum of partial currents due to O<sub>2</sub> generation and H<sub>2</sub>O generation (green), and due to O<sub>2</sub> generation, H<sub>2</sub>O generation and CO<sub>2</sub> generation (red). Bottom panel: Difference between the measured current and the sum of partial currents due to O<sub>2</sub> generation, H<sub>2</sub>O generation, and CO<sub>2</sub> generation with the y-axis having the same scale as for the top panel.

the voltage dependence of H<sub>2</sub>O mass signal to minimize the difference. Now the sum (green line) of the partial current due to O<sub>2</sub> generation (blue line) and that of H<sub>2</sub>O generation (orange line) is in good agreement with the measured current in Region II. Thus, the decline of O<sub>2</sub> generation efficiency after the 1st maximum (in Region II) is well explained by the current used for the generation of H<sub>2</sub>O.

In Region III, the difference between the measured current and the sum of partial current due to O<sub>2</sub> generation and H<sub>2</sub>O generation became apparent. One reason is the delay of mass detection but even the peak height of the sum of the partial currents was smaller than the measured current. The discrepancy was minimized by taking into account the partial current due to CO<sub>2</sub> generation (gray line), which was evaluated by utilizing the voltage dependence of the CO<sub>2</sub> mass signal to minimize the difference. The sum (red line) of the partial currents due to O<sub>2</sub> generation (blue line), H<sub>2</sub>O generation (orange line), and CO<sub>2</sub> generation (gray line) is now in reasonable agreement with the measured current in Region III up to 4.66 V.

In Region IV or at higher voltages (more positive potentials of KJCNT electrode), the contributions of O<sub>2</sub>, H<sub>2</sub>O, and CO<sub>2</sub> generation to current were minimal and the current was dominated by direct anodic decomposition of electrolyte solution as mentioned before.

Thus, anodic current is solely due to O<sub>2</sub> generation by Li<sub>2</sub>O<sub>2</sub> oxidation in Region I, O<sub>2</sub> generation by Li<sub>2</sub>O<sub>2</sub> oxidation and H<sub>2</sub>O generation possibly by LiOH oxidation in Region II, O<sub>2</sub> generation mainly by Li<sub>2</sub>O<sub>2</sub> oxidation, H<sub>2</sub>O generation possibly by LiOH oxidation, and CO<sub>2</sub> generation mainly by Li<sub>2</sub>CO<sub>3</sub> oxidation in Region III, and direct anodic oxidation of electrolyte solution in Region IV.



## Behaviours of current and mass signals during charge by potential steps

To see how the generation of  $O_2$ ,  $H_2O$ , and  $CO_2$  proceeded, voltage step experiments were also performed.

Fig. 5 shows the responses of current (middle panel) and signal intensities of  $m/z = 18$  (orange), 32 (blue), and 44 (gray) when voltage steps were applied with the sequence shown in the bottom panel (300 mV voltage step and 6000 s stay). Since the background signal of  $m/z = 18$  ( $H_2O$ ) was rather high ( $\sim 1.5 \times 10^{-11} A$ ), the data shown in the figure are background corrected.

When the voltage was stepped from 2.9 V to 3.2 V (within Region I), 3.2 V to 3.5 V (Region I to Region II), 3.5 V to 3.8 V (within Region II), and 3.8 V to 4.1 V (within Region II) and kept at a given value for 6000 s, the current increased sharply and declined monotonously with time. The time dependence of mass signal for  $O_2$  ( $m/z = 32$ ) was similar to that of current in these cases, although there existed some delay in the rising portion of oxygen generation compared to the current rise. It is interesting to note that both current and  $O_2$  signal became small after 6000 s but they increased significantly again when the next potential step was applied. This means that there existed  $Li_2O_2$  which cannot be oxidized at a given potential, for example at 3.5 V, but can be oxidized at higher voltages, for example at 3.8 V. The increase of the  $H_2O$  signal ( $m/z = 18$ ) was also observed when these voltage steps were applied except for the voltage step from 2.9 V to 3.2 V (within Region I) as expected from the LVS (Fig. 3). The time dependence of the  $H_2O$  signal was, however, quite different from those of the current and  $O_2$  signal. It increased very slowly upon the voltage step and declined very slightly after a peak. No significant change in the

$CO_2$  signal ( $m/z = 44$ ) was observed when these voltage steps were applied as expected from the LVS (Fig. 3).

When the voltage was stepped from 4.1 V to 4.4 V (Region II to Region III), the current-time relationship was quite different from those observed in other voltage step cases reported above. After the current spike due to double layer charging, the current increased and then decreased again but not monotonously, showing rather complex processes took place in this case. The behaviour of the  $O_2$  signal was also quite different from those observed in other voltage step cases. Although it increased upon potential step and then declined as in other potential step cases, the decline was not monotonous but with a shoulder. The time dependencies of the current and the  $O_2$  signal can be deconvoluted with two waves; initial oxidation of  $l-Li_2O_2$  (simple decay) followed by the oxidation of  $h-Li_2O_2$  (complex curve). Thus, while only  $l-Li_2O_2$  was oxidized when the voltage step was up to 4.1 V, both  $l-Li_2O_2$  and  $h-Li_2O_2$  were oxidized, first  $l-Li_2O_2$  oxidation followed by  $h-Li_2O_2$  oxidation, when the voltage was stepped to 4.4 V, corresponding to the 1st and the 2nd peak in the LVS (Fig. 3). These results support the conclusion drawn from LVS measurement that the natures (energetics) and the oxidation mechanisms of  $l-Li_2O_2$  and  $h-Li_2O_2$  were different. The behaviour of the  $m/z = 18$  ( $H_2O$ ) signal was similar to those observed in other voltage step cases as it increased very slowly upon the voltage step and declined very slightly after a peak. The  $m/z = 44$  ( $CO_2$ ) signal increased corresponding to this voltage step but in a rather complex way and declined after a peak as in the  $h-Li_2O_2$  oxidation. This behaviour shows that the generation of  $CO_2$  was associated with  $h-Li_2O_2$  oxidation.

The presence of two different types of  $Li_2O_2$  and the relationship between  $Li_2O_2$  oxidation ( $O_2$  generation) and the generation of  $H_2O$  and  $CO_2$  can be clearly seen when voltage was stepped from OCP (2.8 V) to 4.4 V, which was close to the 2nd maximum of the signal of  $m/z = 32$  ( $O_2$ ) observed in the LVS (Fig. 3), as shown in Fig. 6.

The current increased sharply and declined but then reached a plateau at *ca.* 0.9 h, and then decreased again.

The signal of  $m/z = 32$  ( $O_2$ ) increased upon voltage step but much more slowly compared with the current, reached a maximum at *ca.* 0.2 h, then declined, reached a plateau at *ca.* 1 h, and then decreased again. This behaviour can be understood by considering the oxidation of  $Li_2O_2$  corresponding to the 1<sup>st</sup> and 2<sup>nd</sup> peaks in the LVS (Fig. 3). The time dependence of the signal  $m/z = 32$  ( $O_2$ ) can be deconvoluted into two peaks, the first peak being the one with a maximum at *ca.* 0.2 h and the 2<sup>nd</sup> peak represented by a Gaussian curve with a maximum at *ca.* 1.2 h as shown by dotted lines (black for the first peak and red for the second peak) in Fig. 6, although more accurate fitting requires the contribution of oxygen generation accompanied with the generation of  $H_2O$  (*e.g.*,  $LiOH \rightarrow Li^+ + \frac{1}{2}H_2O + \frac{1}{4}O_2 + e^-$ : *ca.* 3.4 V) and  $CO_2$  (*e.g.*,  $Li_2CO_3 \rightarrow 2Li^+ + CO_2 + \frac{1}{2}O_2 + 2e^-$ : *ca.* 3.8 V). The integrated amount of the 2<sup>nd</sup> oxygen peak based on the fitting was in good agreement with that obtained in the LVS as shown in Table 1.

Thus, it is confirmed that there existed two types of  $Li_2O_2$ , one oxidized more easily than the other. The first type of  $Li_2O_2$

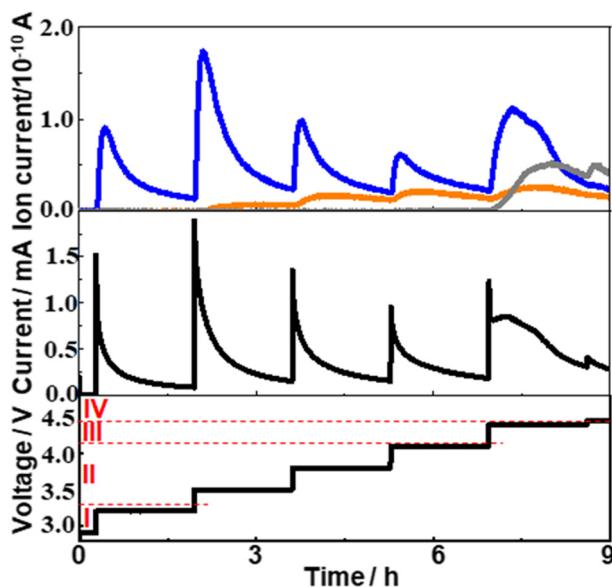


Fig. 5 Time dependencies of current (middle panel) and mass signal of  $m/z = 18$  (orange), 32 (blue), and 44 (gray) (top panel) when the sequences of the voltage step (300 mV step and 6000 s stay before the next step) were applied as shown in the bottom panel.



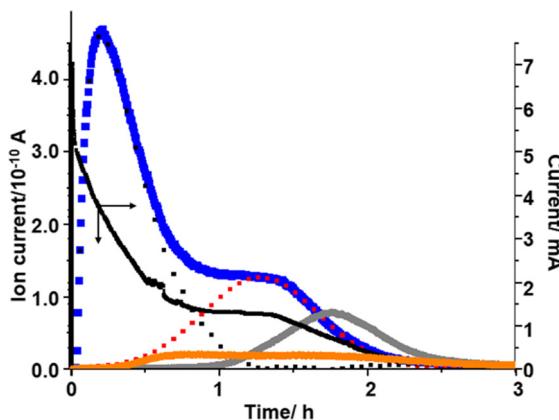


Fig. 6 Time dependences of current (black) and signals of  $m/z = 18$  (orange), 32 (blue), and 44 (gray) when the voltage was stepped from 2.8 V to 4.4 V. The red dotted line shows Gaussian fitting for the 2nd oxygen generation and the black dotted line is the residue for  $O_2$  generation after the subtraction of the red curve.

(l- $Li_2O_2$ ) is oxidized at lower voltages in the LVS (Fig. 3) and in sequential step (Fig. 5) modes, and earlier when the voltage was stepped to and kept at 4.4 V where both types of  $Li_2O_2$  can be oxidized (Fig. 6). These results also show that  $Li_2O_2$ , which corresponded to the 2nd oxygen peak in the LVS (h- $Li_2O_2$ ), was oxidized after the oxidation of  $Li_2O_2$ , which corresponded to the 1st oxygen peak in the LVS (h- $Li_2O_2$ ), which was completed in all charge modes.

The  $m/z = 18$  ( $H_2O$ ) signal started to increase at around 0.2 h, which was close to the position of the 1st oxygen peak, increased rather slowly, reached a broad maximum at *ca.* 0.75 h, then declined and increased slightly, reached the 2nd maximum at *ca.* 1.6 h, and then decreased slowly. This behaviour is similar to that of  $H_2O$  generation in the LVS and sequential step modes. The integrated amount of mass signal of  $H_2O$  was in reasonable agreement with that obtained in the LVS measurement as also shown in Table 1.

The  $m/z = 44$  ( $CO_2$ ) signal started to increase at around 1 h when the 1st oxygen generation was almost completed, reached a maximum at *ca.* 1.8 h, and then declined. The integrated amounts of the signal of  $m/z = 44$  were in good agreement with that obtained in the LVS measurement as also shown in Table 1. It must be stressed that  $CO_2$  generation started when the oxidation of the first type of  $Li_2O_2$  corresponding to the 1st oxygen peak in the LVS was completed and the second type of  $Li_2O_2$  corresponding to the 2nd oxygen peak in the LVS was

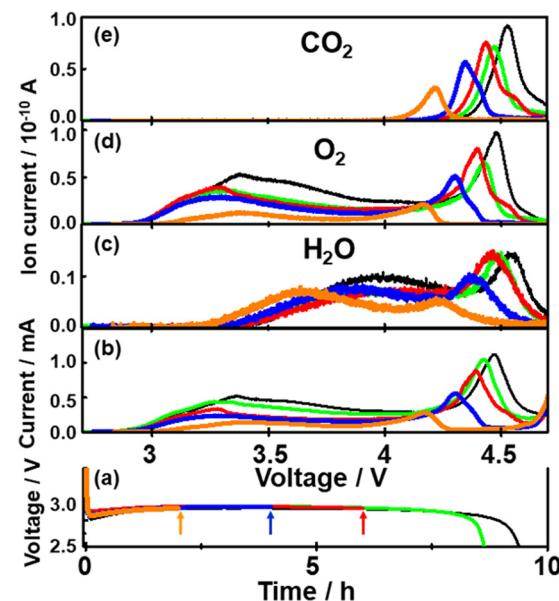


Fig. 7 (a) Discharge curves of various cells with discharge times of 2 (orange), 4 (blue), 6 (red), 8.8 (green), and 9.6 h (black). (b)–(e) Potential dependencies of (b) current and the background corrected ion currents of mass signals of (c)  $m/z = 18$  ( $H_2O$ ), (d) 32 ( $O_2$ ), and (e) 44 ( $CO_2$ ) as a function of time for 2 (orange), 4 (blue), 6 (red), 8.8 (green) and 9.6 h (black) discharged cells.

started as were the cases in the LVS (Fig. 3) and sequential step (Fig. 5) modes.

### 3.4. Effect of discharge time on the behaviours of current and mass signals during LVS charge

To further understand the relationship between  $O_2$  generation ( $Li_2O_2$  oxidation) and the generations of  $H_2O$  and  $CO_2$ , time (voltage) dependencies of current and signal intensities of  $m/z = 18$  ( $H_2O$ ), 32 ( $O_2$ ), and 44 ( $CO_2$ ) during LVS charge ( $0.05\text{ mV s}^{-1}$ ) were obtained after the constant current ( $0.2\text{ mA cm}^{-2}$ ) discharge of various time durations.

Fig. 7 shows the results for the cells with discharge times of 2 (orange), 4 (blue), 6 (red), 8.8 (green), and 9.6 h (black). It is clear that as the discharge time increased, both current and mass signals increased and the positions of the 2nd current peak, the 2nd peaks of mass signals of  $m/z = 18$  and 32, and the peak of mass signal of  $m/z = 44$  shifted to higher voltages.

Fig. 8 shows the integrated amounts of current (=charge) and mass signals of  $m/z = 18$ , 32, and 44, which are normalized by the values of corresponding signals obtained for 9.6 h discharge.

Table 1 Integrated amounts of current (charge) and mass signals corresponding to  $O_2$ ,  $H_2O$ , and  $CO_2$  during the LVS (Fig. 3) and those when the voltage was stepped to 4.4 V (Fig. 6)

Integrated current (charge)/mA h	Integrated mass signal (ion current)/ $10^{-10}\text{ A h}$				
	$m/z = 32$ ( $O_2$ )			$m/z = 18$ ( $H_2O$ )	$m/z = 44$ ( $CO_2$ )
	1st	2nd	Total		
LVS	3.88	2.08	1.17	3.25	0.524
Step to 4.4 V	3.73	2.15	1.12	3.27	0.406

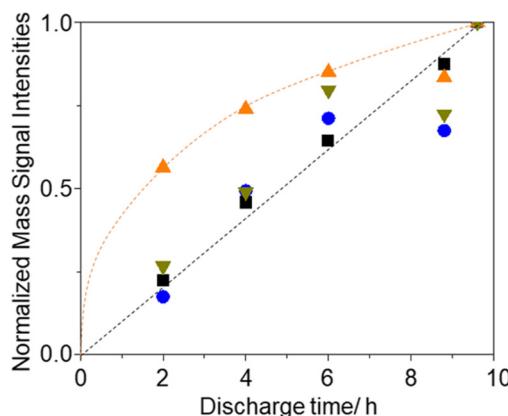


Fig. 8 Discharge time dependencies of integrated signal intensities, *i.e.*, current ( $\blacktriangledown$ ), and ion current of  $m/z = 18$  ( $\blacktriangle$ ),  $32$  ( $\bullet$ ), and  $44$  ( $\blacksquare$ ), normalized by those of  $9.6$  h discharge.

Except for the  $m/z = 18$  ( $\text{H}_2\text{O}$ ) signal, integrated amounts are linearly increased with the discharge time. These results suggest that some species, which were converted to  $\text{H}_2\text{O}$  during charge, *i.e.*,  $\text{LiOH}$ , was preferentially formed in the earlier stage of discharge with residual water as  $\text{LiOH}$  formation is thermodynamically more favoured than  $\text{Li}_2\text{O}_2$  formation if water is present.<sup>66</sup>

Fig. 9 shows the discharge time dependencies of the positions of the second current peak (+), the second peak of  $m/z = 18$  ( $\blacksquare$ ), the second peak of  $m/z = 32$  ( $\bullet$ ), and the peak of  $m/z = 44$  ( $\blacktriangle$ ).

All peaks shifted positively with discharge time as mentioned above. The positions of the peaks are: the second current peak  $\approx$  the second peak of  $m/z = 32$   $<$  the peak of  $m/z = 44$   $<$  the second peak of  $m/z = 18$  in all the cases. The positions of the second peaks of  $m/z = 18$  and  $32$  and the peaks of  $m/z = 44$  are linearly related to the positions of the second current peak for different discharge times (inset of Fig. 9), showing the relative positions of these peaks are constant regardless of discharge time with *ca.*  $60$  mV and  $45$  mV differences between the  $m/z = 32$

and  $18$  signals and  $m/z = 32$  and  $44$  signals, respectively. These results suggest that the generation of  $\text{CO}_2$  proceeded not at potentials more positive than a unique potential such as the redox potential of  $\text{Li}_2\text{CO}_3$  which is about  $3.8$  V,<sup>66,74</sup> but when the oxidation of  $\text{Li}_2\text{O}_2$  corresponding to the  $2$ nd oxygen peak proceeded, supporting the above suggestion that the source of  $\text{CO}_2$ , which is most probably  $\text{Li}_2\text{CO}_3$  generated during discharge, was present at the same sites as the  $\text{Li}_2\text{O}_2$  corresponding to the  $2$ nd oxygen peak.

### 3.5. Source of O atoms in $\text{O}_2$ , $\text{H}_2\text{O}$ , and $\text{CO}_2$ generated during LVS charge evaluated by the effect of two types of isotope oxygen gas used during discharge

An oxygen atom in the charge products,  $\text{O}_2$ ,  $\text{H}_2\text{O}$ , and  $\text{CO}_2$  can be from  $\text{O}_2$  gas, TEGDME, and residual  $\text{H}_2\text{O}$  in the cell during discharge. By flowing the isotope oxygen gas,  $^{16}\text{O}_2$  or  $^{18}\text{O}_2$ , during discharge and analysing the total fractions of  $^{16}\text{O}$  and  $^{18}\text{O}$  isotopes in the charge products,  $\text{O}_2$ ,  $\text{H}_2\text{O}$ , and  $\text{CO}_2$ , and voltage dependencies of the fractions of isotopes in  $\text{O}_2$ ,  $\text{H}_2\text{O}$ , and  $\text{CO}_2$  during charge should provide useful information to understand the details of not only the growth mechanism of  $\text{Li}_2\text{O}_2$  as studied in previous studies,<sup>56,70,71</sup> but also the mechanisms through which the source compounds of the  $\text{H}_2\text{O}$  and  $\text{CO}_2$ , charge products, are generated during discharge.

Fig. 10 shows mass signals (ion currents) corresponding to (iii)  $\text{O}_2$ , (iv)  $\text{H}_2\text{O}$ , and (v)  $\text{CO}_2$ , and (ii) currents for various cells with the constant current discharge of  $0.4$  mA ( $0.2$  mA  $\text{cm}^{-2}$ ) under various isotope  $\text{O}_2$  gas flow conditions as shown in the bottom panel (i). The red, blue, and black lines are of the ion currents (mass signals) related to  $^{16}\text{O}$ ,  $^{18}\text{O}$ , and  $^{16}\text{O} + ^{18}\text{O}$ , respectively. In the case of  $\text{O}_2$ , the sum of the ion currents of  $m/z = 32$  ( $^{16}\text{O}^{16}\text{O}$ ) and half of the ion currents of  $m/z = 34$  ( $^{16}\text{O}^{18}\text{O}$ ) are taken as the ion currents due to the contribution of  $^{16}\text{O}$  in  $\text{O}_2$  and the sum of half of the ion currents of  $m/z = 34$  and the ion currents of  $m/z = 36$  ( $^{18}\text{O}^{18}\text{O}$ ) are taken as the ion currents due to the contribution of  $^{18}\text{O}$  in  $\text{O}_2$ . In the case of  $\text{H}_2\text{O}$ , while the ion currents of  $m/z = 20$  were taken as the ion currents due to  $^{18}\text{O}$  ( $\text{H}_2^{18}\text{O}$ ), the ion currents due to  $^{16}\text{O}$  ( $\text{H}_2^{16}\text{O}$ ) were obtained using the ion currents of  $m/z = 18$  but after the contribution of  $^{18}\text{O}$  was corrected by considering the fragmentation of  $\text{H}_2^{18}\text{O}$  ( $m/z = 20$ ),  $^{16}\text{O}^{18}\text{O}$ ,  $^{18}\text{O}_2$ ,  $\text{C}^{16}\text{O}^{18}\text{O}$ , and  $\text{C}^{18}\text{O}_2$  to  $^{18}\text{O}$  ( $m/z = 18$ ). In the case of  $\text{CO}_2$ , the sum of the ion currents of  $m/z = 44$  ( $\text{C}^{16}\text{O}^{16}\text{O}$ ) and half of the ion currents of  $m/z = 46$  ( $\text{C}^{16}\text{O}^{18}\text{O}$ ) are taken as the ion currents due to  $\text{CO}_2$  with  $^{16}\text{O}$  contribution and the sum of half of the ion currents of  $m/z = 46$  and the ion currents of  $48$  ( $\text{C}^{18}\text{O}^{18}\text{O}$ ) are taken as the ion currents due to  $\text{CO}_2$  with  $^{18}\text{O}$  contribution. The natural abundance of  $^{18}\text{O}$  in  $^{16,18}\text{O}_2$  gas and the concentrations of  $^{16}\text{O}$  impurities in  $^{16,18}\text{O}_2$  were taken into account in all the analysis and, in addition, the natural abundances of  $^{13}\text{C}$  was taken into account for the  $\text{CO}_2$  analysis.

The relationships between the fraction of  $^{18}\text{O}$  in the  $\text{O}_2$  gas flow during discharge and those in  $\text{O}_2$ ,  $\text{H}_2\text{O}$ , and  $\text{CO}_2$  generated during discharge were investigated. Fig. 11 shows the fractions of  $^{18}\text{O}$  in total amounts of  $\text{O}_2$ ,  $\text{H}_2\text{O}$ , and  $\text{CO}_2$  generated during

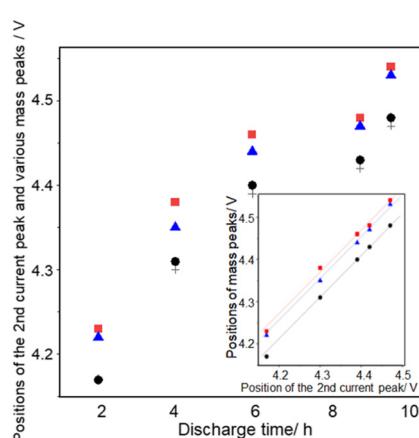


Fig. 9 Positions of the second peak of current (+),  $18$  ( $\blacksquare$ ), and  $32$  ( $\bullet$ ), and the peak of  $44$  ( $\blacktriangle$ ) as a function of discharge time. Inset: Positions of the second peak of  $18$  ( $\blacksquare$ ) and  $32$  ( $\bullet$ ), and the peak of  $44$  ( $\blacktriangle$ ) as a function of the positions of the second current peak.



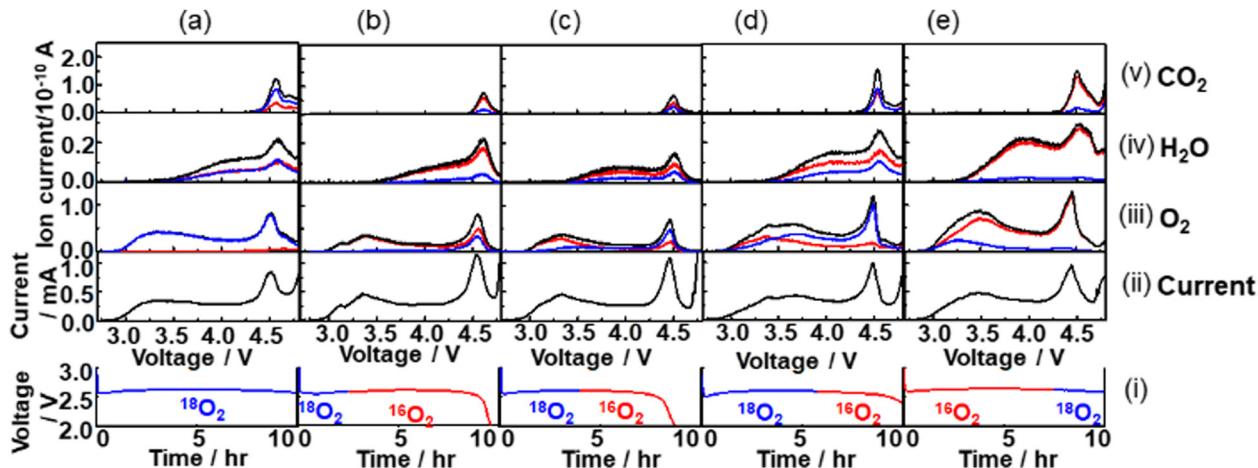


Fig. 10 Mass signals (ion currents) for (iii)  $\text{O}_2$ , (iv)  $\text{H}_2\text{O}$ , and (v)  $\text{CO}_2$ , and (ii) currents for various cells with the constant current discharge of 0.4 mA ( $0.2 \text{ mA cm}^{-2}$ ) under various isotope  $\text{O}_2$  gas flow conditions as shown in the bottom panel (i). (a) 10 h  $^{18}\text{O}_2$  flow, (b) 2.5 h  $^{18}\text{O}_2$  flow followed by 7.1 h  $^{16}\text{O}_2$  flow, (c) 4 h  $^{18}\text{O}_2$  flow followed by 4.6 h  $^{16}\text{O}_2$  flow, (d) 5.8 h  $^{18}\text{O}_2$  flow followed by 4.2 h  $^{16}\text{O}_2$  flow, and (e) 7.5 h  $^{16}\text{O}_2$  flow followed by 2.5 h  $^{18}\text{O}_2$  flow. In (iii), (iv), and (v), red, blue, and black lines are of mass signals related to  $^{16}\text{O}$ ,  $^{18}\text{O}$ , and  $^{16}\text{O} + ^{18}\text{O}$ , respectively.

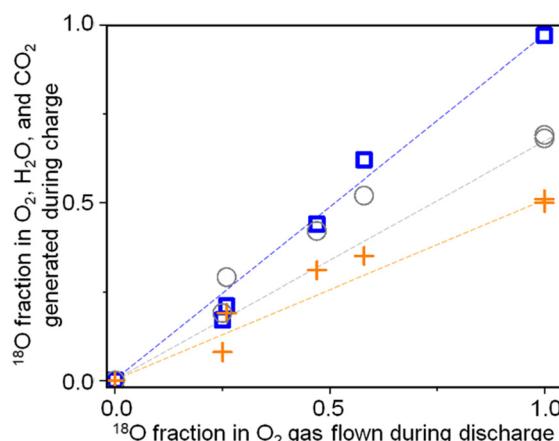


Fig. 11 Fractions of  $^{18}\text{O}$  in  $\text{O}_2$  (□),  $\text{H}_2\text{O}$  (+), and  $\text{CO}_2$  (○) generated during charge as a function of the fraction of  $^{18}\text{O}$  in the  $\text{O}_2$  gas flow during discharge.

charge as a function of the fraction of  $^{18}\text{O}$  in  $\text{O}_2$  gas flow during discharge.

Although they are linearly correlated in all the cases, the slopes are different. While the fractions of  $^{18}\text{O}$  in  $\text{O}_2$  generated during charge are the same as those in the  $\text{O}_2$  gas flow during discharge, those in  $\text{H}_2\text{O}$ , and  $\text{CO}_2$  generated during charge are less than those in the  $\text{O}_2$  gas flow during discharge. Only 50% and 70% of O in  $\text{H}_2\text{O}$  and  $\text{CO}_2$  generated during charge are  $^{18}\text{O}$  even when  $^{18}\text{O}_2$  flowed throughout the discharge (Fig. 10(a)). These results show that while all the O atoms in  $\text{Li}_2\text{O}_2$  originated from the  $\text{O}_2$  gas flow during discharge, around 50% and 30% of the O atoms in the compounds, which were sources of  $\text{H}_2\text{O}$  and  $\text{CO}_2$  during charging, respectively, generated during discharge originated from O atoms of TEGDME. More quantitative discussion on how O atoms were incorporated into the precursors of  $\text{H}_2\text{O}$  and  $\text{CO}_2$ , plausibly  $\text{LiOH}$  and  $\text{Li}_2\text{CO}_3$ , respectively, during discharge is required to fully

understand the degradation mechanism of TEGDME and is now underway.

It is confirmed that all the O atoms in  $\text{Li}_2\text{O}_2$  seemed to originate from the  $\text{O}_2$  gas flow during discharge by the results shown in Fig. 12, where integrated mass signals (ion currents) corresponding to  $\text{O}_2$  (Fig. 10(iii)) with  $^{16}\text{O}$  (red) or  $^{18}\text{O}$  (blue) contributions during charge with an LVS sweep of  $0.05 \text{ mV s}^{-1}$  for various cells are plotted as a function of charge passed

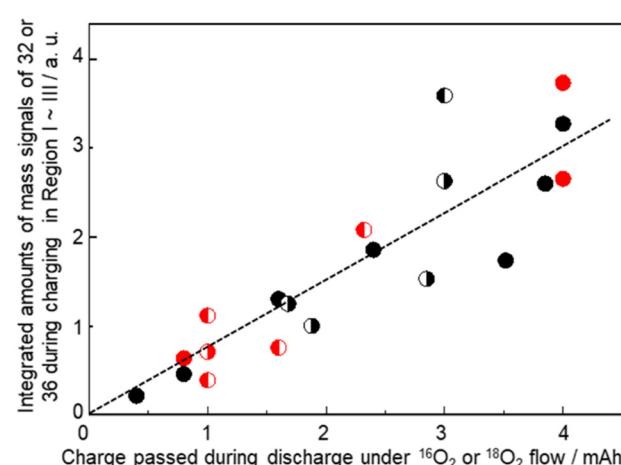


Fig. 12 Integrated mass signals (ion currents) for 32 or 36 during charge with the voltage sweep of  $0.05 \text{ mV s}^{-1}$  for various cells as a function of charge passed during discharge under the corresponding isotope  $\text{O}_2$  gas flow. Black (●) and red (●) closed circles correspond to the cells discharged under a single isotope of  $^{16}\text{O}_2$  gas and  $^{18}\text{O}_2$  gas flow, respectively, throughout the discharge. Half-closed circles correspond to the cells discharged under two isotopes of  $\text{O}_2$  gas in sequence with the filled half for the order of gas flow. ○ and ● are the integrated amounts of  $\text{O}_2$  based on  $^{16}\text{O}$  and  $^{18}\text{O}$ , respectively, during charge for the cells with an initial  $^{16}\text{O}_2$  gas flow followed by  $^{18}\text{O}_2$  gas flow during discharge. ○ and ● are the integrated amounts of  $\text{O}_2$  based on  $^{18}\text{O}$  and  $^{16}\text{O}$ , respectively, during charge for the cells with the initial  $^{18}\text{O}_2$  gas flow followed by  $^{16}\text{O}_2$  gas flow during discharge.

during discharge under the corresponding isotope  $O_2$  gas flow. Black (●) and red (●) closed circles are for the cells discharged under  $O_2$  gas of a single isotope *i.e.*,  $^{16}O_2$  gas and  $^{18}O_2$  gas flow, respectively, throughout the discharge. Data of discharge time dependencies under  $^{16}O_2$  gas flow (Fig. 7 and 9) and  $^{18}O_2$  gas flow (Fig. 10) are included. Half-closed circles correspond to the cells discharged under two isotopes of  $O_2$  gas in sequence with the filled half for the order of gas flow. Thus, ● and ● are the integrated amounts of  $O_2$  based on  $^{16}O$  and  $^{18}O$ , respectively, during the charge for the cells with the initial  $^{16}O_2$  gas flow followed by  $^{18}O_2$  gas flow during discharge, and ● and ● are the integrated amounts of  $O_2$  based on  $^{18}O$  and  $^{16}O$ , respectively, during the charge for the cells with the initial  $^{18}O_2$  gas flow followed by  $^{16}O_2$  gas flow during discharge. Although the data are scattered, the amounts of  $O_2$  based on  $^{16}O$  and  $^{18}O$  are linearly related to the charge passed during discharge under  $^{16}O_2$  and  $^{18}O_2$  gas flow, respectively, regardless of the order of the gas flow. Thus,  $Li_2O_2$  is formed with  $O_2$ , which flowed at the time of  $Li_2O_2$  formation.

**Voltage dependencies of the fraction of isotopes in  $O_2$  generated during charge.** Although the features of time dependencies of total  $O_2$  generations are essentially the same in all the cases, those of the contributions of individual isotopes ( $^{16}O$  and  $^{18}O$ ) were strongly dependent on the way how  $^{16}O_2$  and  $^{18}O_2$  were supplied to the cell during the discharge. It is clear that the  $O$  isotope, initially introduced during discharge, was released later during charge. If  $^{18}O_2$  was initially supplied to the cell followed by  $^{16}O_2$  as shown in the bottom panel (i) of Fig. 10(b-d),  $^{16}O_2$  and  $^{18}O_2$  were dominant in  $O_2$  gas generated in the earlier and later stages of charge, respectively. For example, when  $^{18}O_2$  was initially supplied to the cell for 2.5 h followed by  $^{16}O_2$  for 7.1 h (Fig. 10(b)-(i)),  $^{18}O_2$  was detected only in the 2nd oxygen peak and did not contribute to the 1st oxygen peak (Fig. 10(b)-(iii)). The longer the initial  $^{18}O_2$  supply during discharge, the higher the contribution of  $^{18}O_2$  during charge, but the trend that  $^{16}O_2$  and  $^{18}O_2$  were dominant in  $O_2$  gas generated in the earlier and later stages of charge, respectively, was valid (Fig. 10(c) and (d)). Similarly, if  $^{16}O_2$  was initially supplied to the cell followed by  $^{18}O_2$  supply as shown in Fig. 10(e)-(i), the reverse trend was observed as  $^{18}O_2$  and  $^{16}O_2$  were dominant in  $O_2$  gas generated in the earlier and later stages of charge, respectively (Fig. 10(e)-(iii)).

The trends can be more clearly seen in Fig. 13 where the fraction of the  $^{18}O$  isotope, which was contained in the initial flow of  $O_2$  gas during discharge, and in  $O_2$  gas generated during charging is plotted as a function of cell voltage during LVS charging. In all the cases, the fraction was lowest at the beginning of the LVS charge and increased as the voltage increased up to *ca.* 4.5 V which corresponded to the 2nd current/ $O_2$  peak, confirming that the  $O$  isotope initially introduced during discharge was released in the later stage of charge. Behaviours in higher charge voltages were dependent on the order of isotope supply. When  $^{18}O_2$  was supplied initially during discharge, the fraction of the initially introduced  $O$  isotope ( $^{18}O$ ) reached a maximum at around 4.5 V and decreased at higher voltages (Fig. 13(a)-(c)). On the other hand,

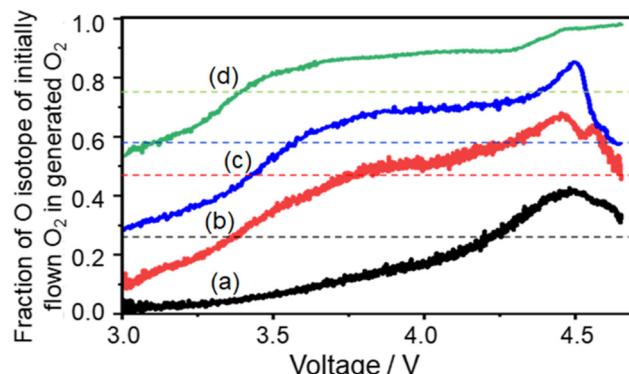


Fig. 13 Fraction of the  $O$  isotope, which was contained in the initially flowing  $O_2$  gas during discharge, and in the  $O_2$  gas generated during LVS charge as a function of cell voltage. (a) 2.5 h  $^{18}O_2$  flow followed by 7.1 h  $^{16}O_2$  flow (black line: data taken from Fig. 10(b)), (b) 4 h  $^{18}O_2$  flow followed by 4.6 h  $^{16}O_2$  flow (red line: data taken from Fig. 10(c)), (c) 5.8 h  $^{18}O_2$  flow followed by 4.2 h  $^{16}O_2$  flow (blue line: data taken from Fig. 10(d)), and (d) 7.5 h  $^{16}O_2$  flow followed by 2.5 h  $^{18}O_2$  flow (green line: data taken from Fig. 10(e)). Dotted lines show the total fractions of the  $^{18}O$  isotope, which was contained in the initially flown  $O_2$  gas during discharge, for whole discharge processes. For example, the black dotted line shows 0.26 (= 2.5/(2.5 + 7.1)) for (a).

when  $^{16}O_2$  was supplied initially during discharge, the fraction of the initially introduced  $O$  isotope ( $^{16}O$ ) increased further when the voltage became larger than 4.5 V (Fig. 13(d)). These results show that the fraction of  $^{16}O$  increased more at voltages larger than *ca.* 4.5 V regardless of the order of the supply of two isotope  $O_2$  gases, suggesting that the source of  $O_2$  generated at larger voltages was not only  $Li_2O_2$ , which is the major discharge product and  $O$  source of which is  $O_2$  gas flowing during discharge, but also other molecules, which were formed during discharge and the  $O$  source of which includes TEGDME. One possible molecule is  $Li_2CO_3$ , which contains  $O$  from TEGDME as mentioned above and generates  $O_2$  accompanied with  $CO_2$  during charge as:  $Li_2CO_3 \rightarrow 2Li^+ + CO_2 + 1/2O_2 + 2e^-$ .

## 4. Conclusions

In the present study, we monitored the generated gaseous products during charging under LVS and the voltage step of LOBs with 1 M LiTFSI TEGDME electrolyte solution in real time by using online mass spectrometry after the cell was discharged under constant current with a continuous flow of  $^{16}O_2$  or  $^{18}O_2$  gas and various discharge times. Based on the results obtained in the present study, we can conclude the following.

(1) At least two types of  $Li_2O_2$  are formed during discharge. While the  $Li_2O_2$  grown in the later stage of discharge is oxidized in a wide range of relatively low cell voltages (l- $Li_2O_2$ : 2.8 – *ca.* 4.1 V, depending on the discharge time), the  $Li_2O_2$  grown in the earlier stage of discharge is oxidized in a relatively narrow range of high cell voltages (h- $Li_2O_2$ ) after the l- $Li_2O_2$  is oxidized. Thus, l- $Li_2O_2$  is energetically distributed very widely but the energetic states of h- $Li_2O_2$  are rather sharp, showing that l- $Li_2O_2$  is amorphous or crystalline with low crystallinity and



h-Li<sub>2</sub>O<sub>2</sub> is crystalline with high crystallinity as suggested before.<sup>73</sup>

(2) The current efficiency for the O<sub>2</sub> generation by Li<sub>2</sub>O<sub>2</sub> oxidation during LVS charge varied with voltage. It started to decrease as H<sub>2</sub>O generation started at around 3.36 V and decreased further as CO<sub>2</sub> generation started at around 4.15 V. The total measured current can be well explained by considering partial currents due to O<sub>2</sub> generation by Li<sub>2</sub>O<sub>2</sub> oxidation, H<sub>2</sub>O generation, and CO<sub>2</sub> generation up to *ca.* 4.66 V. In a higher voltage region, currents due to O<sub>2</sub>, H<sub>2</sub>O, and CO<sub>2</sub> generation became small and those due to the decomposition of TEGDME and its degraded compounds increased significantly.

(3) The most plausible origin of H<sub>2</sub>O is the oxidation of LiOH, which is formed during discharge. Only about 50% of O in H<sub>2</sub>O generated during the charge comes from O<sub>2</sub> gas flown during discharge. The remaining 50% O in H<sub>2</sub>O should originate from TEGDME, although there is a possibility that some fraction of O comes from residual H<sub>2</sub>O in the electrolyte solution.

(4) CO<sub>2</sub> generation during LVS charge started not at a unique voltage but at a voltage, where O<sub>2</sub> generation due to h-Li<sub>2</sub>O<sub>2</sub> oxidation started, depending on the discharge time. Once h-Li<sub>2</sub>O<sub>2</sub> oxidation was completed, almost no CO<sub>2</sub> was generated even when higher voltages were applied, although the generation of organic molecules due to the degradation of the electrolyte, *i.e.*, TEGDME, increased, showing CO<sub>2</sub> generation during LVS charge is not due to the oxidation of TEGDME and/or its degraded compounds but due to the oxidation of Li<sub>2</sub>CO<sub>3</sub>, which is formed during discharge and is present at the same sites of the h-Li<sub>2</sub>O<sub>2</sub>. Only about 70% of O in CO<sub>2</sub> generated during the charging comes from the O<sub>2</sub> gas flow during discharge and the other fraction of O should originate from TEGDME.

More detailed discussion on the mechanisms of TEGDME degradation and its relationship to the generation of precursors of H<sub>2</sub>O and CO<sub>2</sub>, plausibly LiOH and Li<sub>2</sub>CO<sub>3</sub>, respectively, during discharge is underway.

## Conflicts of interest

There are no conflicts to declare.

## Acknowledgements

YG is a NIMS Junior Research Fellow of the Center for Advanced Battery Collaboration supported by JST Grant (JPMJPF2016). KJCNT sheets were donated by KJCNT Specialty Paper. Characterizations of materials and cell assembly were carried out in a NIMS Battery Research Platform.

## Notes and references

1 M. M. Thackeray, C. Wolverton and E. D. Isaacs, *Energy Environ. Sci.*, 2012, **5**, 7854–7863.

- 2 K. Kanamura, Large-scale Batteries for Green Energy Society, in *Electrochemical Science for a Sustainable Society*, ed. K. Uosaki, Springer, 2017, pp. 175–195.
- 3 M. Fichtner, K. Edström, E. Ayerbe, M. Berecibar, A. Bhowmik, I. E. Castelli, S. Clark, R. Dominko, M. Erakca and A. A. Franco, *Adv. Energy Mater.*, 2022, **12**, 2102904.
- 4 M. Li, J. Lu, Z. Chen and K. Amine, *Adv. Mater.*, 2018, **30**, 1800561.
- 5 P. G. Bruce, S. A. Freunberger, L. J. Hardwick and J.-M. Tarascon, *Nat. Mater.*, 2012, **11**, 19–29.
- 6 *Principles and Applications of Lithium Secondary Batteries*, ed. J.-K. Park, John Wiley & Sons, 2012, pp. 1–7.
- 7 M. Winter, B. Barnett and K. Xu, *Chem. Rev.*, 2018, **118**, 11433–11456.
- 8 P. Canepa, G. Sai Gautam, D. C. Hannah, R. Malik, M. Liu, K. G. Gallagher, K. A. Persson and G. Ceder, *Chem. Rev.*, 2017, **117**, 4287–4341.
- 9 T. Placke, R. Kloepsch, S. Dühnen and M. Winter, *J. Solid State Electrochem.*, 2017, **21**, 1939–1964.
- 10 M. Ue, K. Sakaushi and K. Uosaki, *Mater. Horiz.*, 2020, **7**, 1937–1954.
- 11 N. Imanishi, A. C. Luntz and P. Bruce, *The Lithium Air Battery: Fundamentals*, Springer, 2014, pp. 1–21.
- 12 K. Abraham and Z. Jiang, *J. Electrochem. Soc.*, 1996, **143**, 1.
- 13 S. Matsuda, M. Ono, S. Yamaguchi and K. Uosaki, *Mater. Horiz.*, 2022, **9**, 856–863.
- 14 N. B. Aetukuri, B. D. McCloskey, J. M. García, L. E. Krupp, V. Viswanathan and A. C. Luntz, *Nat. Chem.*, 2015, **7**, 50–56.
- 15 W.-J. Kwak, Rosy, D. Sharon, C. Xia, H. Kim, L. R. Johnson, P. G. Bruce, L. F. Nazar, Y.-K. Sun and A. A. Frimer, *Chem. Rev.*, 2020, **120**, 6626–6683.
- 16 Z. Wu, Y. Tian, H. Chen, L. Wang, S. Qian, T. Wu, S. Zhang and J. Lu, *Chem. Soc. Rev.*, 2022, **51**, 8045–8101.
- 17 G. Girishkumar, B. McCloskey, A. C. Luntz, S. Swanson and W. Wilcke, *J. Phys. Chem.*, 2010, **1**, 2193–2203.
- 18 J. P. Zheng, R. Y. Liang, M. Hendrickson and E. J. Plichta, *J. Electrochem. Soc.*, 2008, **155**, A432.
- 19 Z. Liang, W. Wang and Y.-C. Lu, *Joule*, 2022, **6**, 2458–2473.
- 20 A. C. Luntz and B. D. McCloskey, *Chem. Rev.*, 2014, **114**, 11721–11750.
- 21 H.-G. Jung, J. Hassoun, J.-B. Park, Y.-K. Sun and B. Scrosati, *Nat. Chem.*, 2012, **4**, 579–585.
- 22 H. Woo, J. Kang, J. Kim, C. Kim, S. Nam and B. Park, *Electron. Mater. Lett.*, 2016, **12**, 551–567.
- 23 S. B. Ma, D. J. Lee, V. Roe, D. Im and S.-G. Doo, *J. Power Sources*, 2013, **244**, 494–498.
- 24 D. Qu, *AIP Conference Proceedings, Fundamental Principles of Battery Design: Porous Electrodes*, USA, 2014, vol. 07.
- 25 J. Xiao, D. Mei, X. Li, W. Xu, D. Wang, G. L. Graff, W. D. Bennett, Z. Nie, L. V. Saraf and I. A. Aksay, *Nano Lett.*, 2011, **11**, 5071–5078.
- 26 L. Liu, Y. Liu, C. Wang, X. Peng, W. Fang, Y. Hou, J. Wang, J. Ye and Y. Wu, *Small Methods*, 2022, **6**, 2101280.
- 27 F. Wang, X. Li, Y. Xie, Q. Lai and J. Tan, *ACS Appl. Energy Mater.*, 2022, **5**, 5473–5483.



28 C. Tran, X.-Q. Yang and D. Qu, *J. Power Sources*, 2010, **195**, 2057–2063.

29 P. Albertus, T. Lohmann and J. Christensen, Overview of LiO<sub>2</sub> Battery Systems with a Focus on Oxygen Handling Requirements and Technologies, in *The Lithium Air Battery: Fundamentals*, ed. N. Imanishi, A. C. Luntz and P. Bruce, Springer, 2014, pp. 306–308.

30 J.-H. Kang, J. Lee, J.-W. Jung, J. Park, T. Jang, H.-S. Kim, J.-S. Nam, H. Lim, K. R. Yoon, W.-H. Ryu, I.-D. Kim and H. R. Byon, *ACS Nano*, 2020, **14**, 14549–14578.

31 D. Geng, N. Ding, T. S. A. Hor, S. W. Chien, Z. Liu, D. Wuu, X. Sun and Y. Zong, *Adv. Energy Mater.*, 2016, **6**, 1502164.

32 N. Imanishi and O. Yamamoto, *Mater. Today*, 2014, **17**, 24–30.

33 J.-G. Zhang, D. Wang, W. Xu, J. Xiao and R. E. Williford, *J. Power Sources*, 2010, **195**, 4332–4337.

34 R. Black, B. Adams and L. Nazar, *Adv. Energy Mater.*, 2012, **2**, 801–815.

35 B. D. McCloskey, D. Bethune, R. Shelby, T. Mori, R. Scheffler, A. Speidel, M. Sherwood and A. Luntz, *J. Phys. Chem.*, 2012, **3**, 3043–3047.

36 Y. Shao, F. Ding, J. Xiao, J. Zhang, W. Xu, S. Park, J. G. Zhang, Y. Wang and J. Liu, *Adv. Funct. Mater.*, 2013, **23**, 987–1004.

37 X. Yao, Q. Dong, Q. Cheng and D. Wang, *Angew. Chem., Int. Ed.*, 2016, **55**, 11344–11353.

38 C. O. Laoire, S. Mukerjee, K. Abraham, E. J. Plichta and M. A. Hendrickson, *J. Phys. Chem. C*, 2010, **114**, 9178–9186.

39 T. Liu, J. P. Vivek, E. W. Zhao, J. Lei, N. Garcia-Araez and C. P. Grey, *Chem. Rev.*, 2020, **120**, 6558–6625.

40 M. Ue and K. Uosaki, *Curr. Opin. Electrochem.*, 2019, **17**, 106–113.

41 F. Li, T. Zhang and H. Zhou, *Energy Environ. Sci.*, 2013, **6**, 1125–1141.

42 W. Xu, K. Xu, V. V. Viswanathan, S. A. Towne, J. S. Hardy, J. Xiao, Z. Nie, D. Hu, D. Wang and J.-G. Zhang, *J. Power Sources*, 2011, **196**, 9631–9639.

43 Y.-C. Lu, B. M. Gallant, D. G. Kwabi, J. R. Harding, R. R. Mitchell, M. S. Whittingham and Y. Shao-Horn, *Energy Environ. Sci.*, 2013, **6**, 750–768.

44 F. S. Gittleson, K. P. Yao, D. G. Kwabi, S. Y. Sayed, W. H. Ryu, Y. Shao-Horn and A. D. Taylor, *ChemElectroChem*, 2015, **2**, 1446–1457.

45 K. Tomita, H. Noguchi and K. Uosaki, *ACS Appl. Energy Mater.*, 2018, **1**, 3434–3442.

46 C. Song, K. Ito, O. Sakata and Y. Kubo, *RSC Adv.*, 2018, **8**, 26293–26299.

47 F. Marchini, S. Herrera, W. Torres, A. Y. Tesio, F. J. Williams and E. J. Calvo, *Langmuir*, 2015, **31**, 9236–9245.

48 K. P. Yao, D. G. Kwabi, R. A. Quinlan, A. N. Mansour, A. Grimaud, Y.-L. Lee, Y.-C. Lu and Y. Shao-Horn, *J. Electrochem. Soc.*, 2013, **160**, A824.

49 B. D. McCloskey, D. S. Bethune, R. M. Shelby, G. Girishkumar and A. C. Luntz, *J. Phys. Chem.*, 2011, **2**, 1161–1166.

50 M. Ue, H. Asahina, S. Matsuda and K. Uosaki, *RSC Adv.*, 2020, **10**, 42971–42982.

51 B. D. McCloskey, R. Scheffler, A. Speidel, D. S. Bethune, R. M. Shelby and A. C. Luntz, *J. Am. Chem. Soc.*, 2011, **133**, 18038–18041.

52 T. Herl and F. M. Matysik, *ChemElectroChem*, 2020, **7**, 2498–2512.

53 X. Xin, K. Ito and Y. Kubo, *ACS Appl. Mater. Interfaces*, 2017, **9**, 25976–25984.

54 Y. Gao, H. Noguchi and K. Uosaki, *RSC Adv.*, 2023, **13**, 5467–5472.

55 Y. Gao, H. Noguchi and K. Uosaki, *ACS Energy Lett.*, 2023, **8**, 1811–1817.

56 K. Tomita, H. Noguchi and K. Uosaki, *J. Am. Chem. Soc.*, 2020, **142**, 19502–19509.

57 K. Nishioka, M. Tanaka, H. Fujimoto, T. Amaya, S. Ogoshi, M. Tobisu and S. Nakanishi, *Angew. Chem., Int. Ed.*, 2022, **61**, e202112769.

58 Z. Zhang, X. Xiao, W. Yu, Z. Zhao and P. Tan, *Nano Lett.*, 2022, **22**, 7527–7534.

59 C. Tan, D. Cao, L. Zheng, Y. Shen, L. Chen and Y. Chen, *J. Am. Chem. Soc.*, 2022, **144**, 807–815.

60 Z. Zhao, J. Huang and Z. Peng, *Angew. Chem., Int. Ed.*, 2018, **57**, 3874–3886.

61 M. M. Ottakam Thotiyil, S. A. Freunberger, Z. Peng and P. G. Bruce, *J. Am. Chem. Soc.*, 2013, **135**, 494–500.

62 M. Balaish, J. W. Jung, I. D. Kim and Y. Ein-Eli, *Adv. Funct. Mater.*, 2020, **30**, 1808303.

63 B. D. McCloskey, A. Speidel, R. Scheffler, D. Miller, V. Viswanathan, J. Hummelshøj, J. Nørskov and A. Luntz, *J. Phys. Chem.*, 2012, **3**, 997–1001.

64 Y. Chen, S. A. Freunberger, Z. Peng, F. Bardé and P. G. Bruce, *J. Am. Chem. Soc.*, 2012, **134**, 7952–7957.

65 S. A. Freunberger, Y. Chen, N. E. Drewett, L. J. Hardwick, F. Bardé and P. G. Bruce, *Angew. Chem., Int. Ed.*, 2011, **50**, 8609–8613.

66 S. Meini, N. Tsioiouvaras, K. U. Schwenke, M. Piana, H. Beyer, L. Lange and H. A. Gasteiger, *Phys. Chem. Chem. Phys.*, 2013, **15**, 11478–11493.

67 D. Zhu, L. Zhang, M. Song, X. Wang, R. Mi, H. Liu, J. Mei, L. W. Lau and Y. Chen, *J. Solid State Electrochem.*, 2013, **17**, 2539–2544.

68 D. Oh, K. Virwani, L. Tadesse, M. Jurich, N. Aetukuri, L. E. Thompson, H.-C. Kim and D. S. Bethune, *J. Phys. Chem. C*, 2017, **121**, 1404–1411.

69 Z. Peng, S. A. Freunberger, Y. Chen and P. G. Bruce, *Science*, 2012, **337**, 563–566.

70 K. Nishioka, K. Morimoto, T. Kusumoto, T. Harada, K. Kamiya, Y. Mukouyama and S. Nakanishi, *J. Am. Chem. Soc.*, 2021, **143**, 7394–7401.

71 Y. Wang and Y. C. Lu, *Angew. Chem.*, 2019, **131**, 7036–7040.

72 J. Wang, Y. Zhang, L. Guo, E. Wang and Z. Peng, *Angew. Chem., Int. Ed.*, 2016, **55**, 5201–5205.

73 S. Ganapathy, B. D. Adams, G. Stenou, M. S. Anastasaki, K. Goubitz, X.-F. Miao, L. F. Nazar and M. Wagemaker, *J. Am. Chem. Soc.*, 2014, **136**, 16335–16344.

74 S. E. Renfrew and B. D. McCloskey, *J. Am. Chem. Soc.*, 2017, **139**, 17853–17860.

