



# Fullerene as a probe molecule for single-atom oxygen reduction electrocatalysts†

 Ning Li,<sup>‡a</sup> Kun Guo,<sup>‡\*</sup> Song Lu,<sup>b</sup> Lipiao Bao,<sup>‡</sup> Zhixin Yu,<sup>‡c</sup> and Xing Lu,<sup>‡\*</sup>

 Cite this: *Chem. Commun.*, 2024, 60, 11964

 Received 1st August 2024,  
 Accepted 24th September 2024

DOI: 10.1039/d4cc03901g

rsc.li/chemcomm

**Fullerenes interact positively with many metal-based catalysts via intense electron transfer. Yet, we here revealed that C<sub>60</sub> serves as a probe due to its deactivation of the active sites of single-atom O<sub>2</sub> reduction electrocatalysts. C<sub>60</sub> adsorption to metal atoms creates steric hindrance that restricts the access of O<sub>2</sub> to the active sites.**

Fullerenes are  $\pi$ -conjugated molecular carbon allotropes with a set of fascinating structural and electronic properties that have triggered significant interest from the catalysis community.<sup>1–3</sup> For instance, the abundant native pentagons render fullerenes an unparalleled feedstock to construct intrinsic defect-rich carbon catalysts.<sup>4–6</sup> The metal-like electrical conductivity brings higher fullerene C<sub>96</sub> with superior oxygen reduction reaction (ORR) activity to that of semiconducting C<sub>60</sub> and C<sub>70</sub>.<sup>7</sup> Besides, covalent 2D fullerene networks with suitable band gaps and high carrier mobility are potent photocatalysts for water splitting.<sup>8,9</sup>

Recently, fullerenes as powerful regulators of the localized electronic structures of metal-based catalysts are increasingly being reported. For example, the exceptional redox capability renders C<sub>60</sub> molecules as a robust electron-buffering agent between Cu<sup>0</sup>/Cu<sup>+</sup> and Cu<sup>2+</sup>/Cu<sup>+</sup> couples.<sup>10</sup> Compared to Cu/SiO<sub>2</sub> with dominant Cu<sup>0</sup> species, C<sub>60</sub> can extract electrons from Cu<sup>0</sup> to form C<sub>60</sub><sup>−</sup> and Cu<sup>+</sup>. Meanwhile, C<sub>60</sub><sup>−</sup> can also donate electrons to Cu<sup>2+</sup>, forming C<sub>60</sub> and Cu<sup>+</sup>. As a result, crucial Cu<sup>+</sup> species are stabilized by C<sub>60</sub> to enhance the hydrogenation activity of C<sub>60</sub>-Cu/SiO<sub>2</sub>. In pursuit of the monovalent Cu<sup>+</sup>

species that play a pivotal role in Cu-based catalysts towards CO<sub>2</sub> electroreduction, researchers have devised C<sub>60</sub>-adsorbed Cu metals as superior electrocatalysts to bare Cu.<sup>11,12</sup> Likewise, intense interfacial electron transfer induced by fullerene molecules has also been evidenced to enhance the catalytic performance of metallic substrates, such as CoNi layered double hydroxides<sup>13</sup> and MoS<sub>2</sub>.<sup>14,15</sup> Nevertheless, positive electronic interactions are uncovered only between fullerenes and large-sized metallic aggregates. How sub-nanoscale fullerenes would affect the metal active sites at the atomic level remains an unresolved issue. It is thus of high interest to explore whether fullerenes can still optimize the electronic structure of the metal site where they are adsorbed to, behaving like a feasible axial ligand of metal single-atom catalysts (SACs).<sup>16–21</sup> Or alternatively, the larger sizes of fullerene molecules than that of metal atoms may result in the deactivation of the metal site by surface blocking, functioning as a molecular probe of the active site of SACs.

Macrocycles bearing representative M–N<sub>4</sub>–C moieties have been widely exploited as model catalysts to investigate various carbon-hosted SACs.<sup>22–24</sup> Utilizing metalloporphyrin (MP) as a molecular SAC, we unravel the unexpected inhibiting effect of C<sub>60</sub> in the ORR activity of MP. Control experiments indicate that non-covalently anchoring either cobalt porphyrin (CoP) or C<sub>60</sub> on carbon nanotubes (CoP/CNT and C<sub>60</sub>/CNT) leads to much higher ORR activity than individual CoP, C<sub>60</sub>, and CNTs. However, further addition of C<sub>60</sub> to CoP/CNT (C<sub>60</sub>-CoP/CNT) results in a dramatic ORR activity decay, the same as that observed after adding the well-noted thiocyanate (SCN<sup>−</sup>) poison. Density functional theory (DFT) calculations reveal that the planar MP molecule is confined between the CNT matrix and C<sub>60</sub> molecule, creating high steric hindrance that restricts O<sub>2</sub> access to the active Co atoms.

CoP/CNT and C<sub>60</sub>-CoP/CNT are prepared *via* a straightforward  $\pi$  interaction-enabled adsorption strategy by mixing different precursors in CS<sub>2</sub> solvent (Experimental section, ESI†). In particular, CoP is selected because it presents better ORR activity than the other common 3d metal MPs (Fig. S1, ESI†).<sup>25</sup>

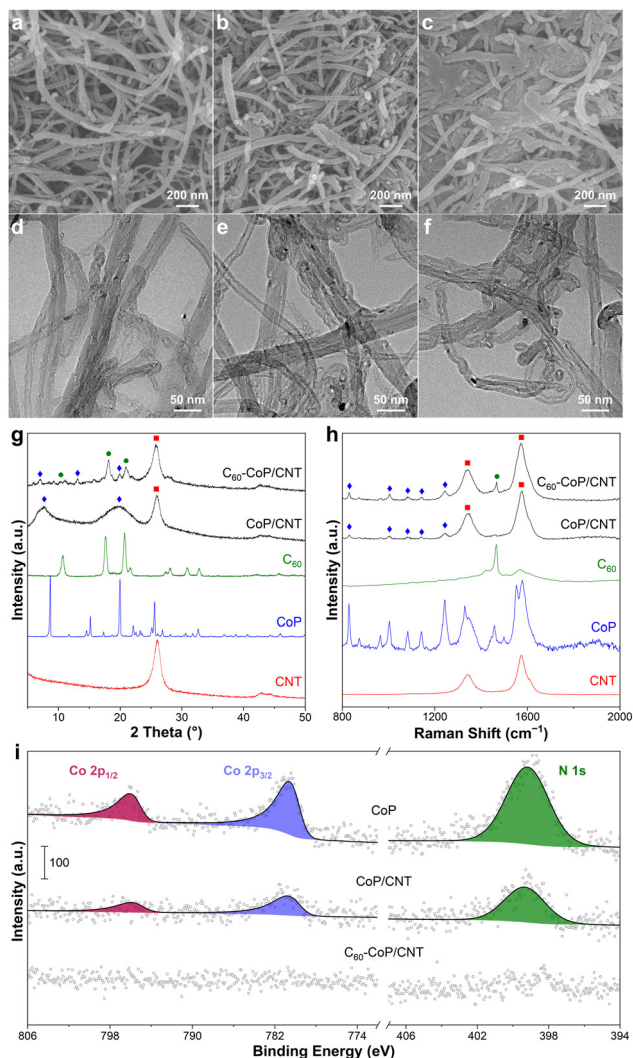
<sup>a</sup> State Key Laboratory of Materials Processing and Die & Mould Technology, School of Materials Science and Engineering, Huazhong University of Science and Technology, Wuhan 430074, China. E-mail: guok@hust.edu.cn, lux@hust.edu.cn

<sup>b</sup> Institute of New Energy, School of Chemistry and Chemical Engineering, Shaoxing University, Shaoxing 312000, China

<sup>c</sup> Department of Energy and Petroleum Engineering, University of Stavanger, Stavanger 4036, Norway

† Electronic supplementary information (ESI) available: Experimental section, characterization and electrochemical data. See DOI: <https://doi.org/10.1039/d4cc03901g>

‡ These authors contributed equally to this work.



**Fig. 1** SEM (a)–(c) and TEM (d)–(f) images of the CNTs (a) and (d), CoP/CNT (b) and (e), and C<sub>60</sub>-CoP/CNT (c) and (f). XRD patterns (g) and Raman spectra (h) of the CNTs, CoP, C<sub>60</sub>, CoP/CNT, and C<sub>60</sub>-CoP/CNT. The red square, blue rhombus, and green circle in (g) and (h) represent CNT, CoP, and C<sub>60</sub>, respectively. (i) XPS spectra of the Co 2p and N 1s regions of CoP, CoP/CNT, and C<sub>60</sub>-CoP/CNT.

The mass loading of CoP is optimized to be 50 wt% due to the highest ORR activity among a series of CoP/CNT (Fig. S2, ESI<sup>†</sup>). The same molar quantity of C<sub>60</sub> as that of CoP is then added in C<sub>60</sub>-CoP/CNT. The morphology and structure of the as-prepared materials are first examined by scanning and transmission electron microscopies (SEM and TEM). The typical 1D morphology is clearly seen in the commercially-received CNTs (Fig. 1a). Despite the high tendency for the self-assembly of CoP and C<sub>60</sub> into large molecular crystals, no obvious aggregates are discerned in either CoP/CNT (Fig. 1b) or C<sub>60</sub>-CoP/CNT (Fig. 1c), which could be attributed to the improved dispersity of both molecules on CNTs. This is further supported by their TEM images. It is found that these CNTs are composed of multi-walled tubes with diameters of dozens of nanometers (Fig. 1d). The uniform image contrasts indicate the absence of

large-sized crystalline particles in CoP/CNT (Fig. 1e) and C<sub>60</sub>-CoP/CNT (Fig. 1f). Additional SEM and TEM images are presented in Fig. S3 (ESI<sup>†</sup>).

The crystallinity of the materials under study is investigated by X-ray diffraction (XRD). As shown in Fig. 1g, a major characteristic peak at  $\sim 26^\circ$ , corresponding to the (002) plane of graphitic carbon, is observed for the CNTs. Pure CoP and C<sub>60</sub> show well-defined characteristic peaks, reflecting the fact that they are easily self-assembled into crystalline particles owing to intermolecular forces.<sup>26</sup> After loading CoP on the CNTs, aside from the diffraction peak of the CNTs, two broad peaks that can be assigned to CoP are discerned, indicating the significantly decreased crystallinity of CoP. A similar phenomenon is also observed after further addition of C<sub>60</sub> to form C<sub>60</sub>-CoP/CNT, in which the C<sub>60</sub> phase presents much lower crystallinity than pure C<sub>60</sub>. The results again manifest that CNTs substantially mitigates the severe aggregation of CoP and C<sub>60</sub> molecules.<sup>27</sup>

Raman spectroscopy is further employed to analyze the chemical structures of these materials. Typical D and G bands of graphitic carbon materials at 1340 and 1570 cm<sup>-1</sup> are observed for CNTs, CoP/CNT, and C<sub>60</sub>-CoP/CNT (Fig. 1h). Besides, multiple weak Raman bands originating from CoP are also observed in CoP/CNT, whereas an additional band at 1460 cm<sup>-1</sup>, namely the A<sub>g</sub>(2) mode of C<sub>60</sub>, is discerned in C<sub>60</sub>-CoP/CNT. The Raman results also confirm that only small-sized CoP and C<sub>60</sub> aggregates are formed after their hybridization with CNTs. Elemental states and speciation information of CoP, CoP/CNT, and C<sub>60</sub>-CoP/CNT are further acquired by X-ray photoelectron spectroscopy (XPS). The full-survey XPS spectra verify the existence of C, N, and Co elements (Fig. S4, ESI<sup>†</sup>). Fig. 1i displays the high-resolution XPS spectra of the Co 2p and N 1s regions, in which the Co 2p spectra can be resolved into two spin-orbit doublets, Co 2p<sub>1/2</sub> and Co 2p<sub>3/2</sub>, whereas the peak centered at 399.2 eV can be assigned to the pyrrolic N of CoP. Compared to CoP, the intensities of the Co 2p<sub>1/2</sub>, Co 2p<sub>3/2</sub>, and N 1s peaks for CoP/CNT become substantially lower, coinciding with the lower mass ratio of CoP in CoP/CNT. Importantly, the Co 2p and N 1s peaks are barely observable for C<sub>60</sub>-CoP/CNT, indicating that the addition of C<sub>60</sub> may lead to the surface coverage of CoP to a certain depth that is beyond the detection limit of XPS.

The ORR performance of the catalysts under study is then evaluated using the standard rotating disk electrode technique in 0.1 M KOH electrolyte (pH = 13). Fig. 2a shows the polarization curves of the ORR on six different catalysts at 1600 rpm. Their corresponding half-wave potentials ( $E_{1/2}$ ) are measured and compared in Fig. 2b. Either C<sub>60</sub> or CNT alone presents rather poor ORR activity with low  $E_{1/2}$  values of 0.55 and 0.64 V, respectively. Despite the presence of Co SACs, CoP still displays limited ORR activity (0.69 V) due probably to severe molecular aggregation and low electrical conductivity. After non-covalently dispersing CoP molecules on conductive CNTs, a significant increase in  $E_{1/2}$  is observed for CoP/CNT (0.80 V), affirming the positive role of the CNT support. This is also supported by loading C<sub>60</sub> onto the CNTs to gain C<sub>60</sub>/CNT, which gives a higher  $E_{1/2}$  of 0.69 V than that of individual C<sub>60</sub> and

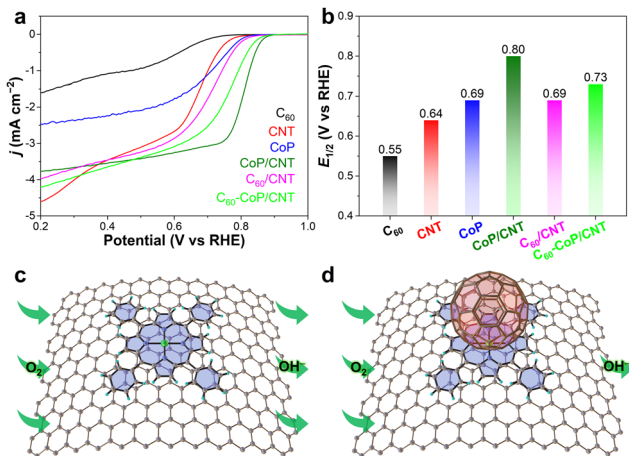


Fig. 2 (a) ORR polarization plots and (b) corresponding  $E_{1/2}$  of the materials under study in 0.1 M KOH at 1600 rpm. Schematic illustrations of CoP/CNT (c) and  $C_{60}$ -CoP/CNT (d) that exhibit high and low ORR activities, respectively. Color code: C, cyan; N, blue; H, white; Co, green.

CNTs, in good line with the results of a previous study.<sup>28</sup> Rotating ring-disk electrode analysis reveals a highly-selective four-electron  $O_2$ -to- $OH^-$  ORR pathway on CoP/CNT (Fig. S5, ESI<sup>†</sup>). More intriguingly, the further addition of  $C_{60}$  to CoP/CNT leads to an unexpected decrease of  $E_{1/2}$  to 0.73 V, indicating an eminent ORR activity decay of Co SACs caused by  $C_{60}$  molecules. Due to the  $\pi$ - $\pi$  stacking interactions, it is no wonder that CoP molecules are planarly adsorbed on the CNT surface, allowing  $O_2$  reactants to be readily adsorbed to and reduced on the exposed Co sites, as illustrated in Fig. 2c. Once  $C_{60}$  molecules are introduced, they could be adsorbed to either CNTs or CoP owing to the analogous  $\pi$ - $\pi$  stacking interactions. Given the deteriorated ORR activity of  $C_{60}$ -CoP/CNT and the aforesaid material characterization results, it is rationalized that  $C_{60}$  shall be situated on top of CoP, potentially blocking the access of  $O_2$  to the ORR-active Co sites (Fig. 2d). On this account,  $C_{60}$  is not an expected facilitator, but rather can serve as a probe molecule to identify the active sites of SACs due to its site deactivation role.

Control experiments are carried out to verify the deactivation role of  $C_{60}$ . We first tune the molar ratio between CoP and  $C_{60}$  (0:1, 1:4, 1:2, and 1:1) of  $C_{60}$ -CoP/CNT. Polarization curves (Fig. 3a) and  $E_{1/2}$  values (Fig. 3b) show that the ORR activity of  $C_{60}$ -CoP/CNT decreases from 0.80 to 0.73 V with the increase of the  $C_{60}$  content. Such positive correlation implies a growing inhibiting effect with more  $C_{60}$  molecules. Moreover,  $SCN^-$  is known as a probe that strongly binds to transition metals, causing chemical, physical, and electronic effects that dramatically disrupt the catalytic functions of such metals.<sup>29–32</sup> It is thus no surprise that the addition of  $SCN^-$  leads to a dramatic activity loss of CoP/CNT (Fig. 3c and d). However, the same  $E_{1/2}$  reduction level by adding  $C_{60}$  as that by adding  $SCN^-$  strongly hints that  $C_{60}$  behaves just like  $SCN^-$  as an inhibitor to the Co SACs, indicating that  $C_{60}$  can serve as a similarly efficient probe to examine the catalytically active sites as  $SCN^-$ .

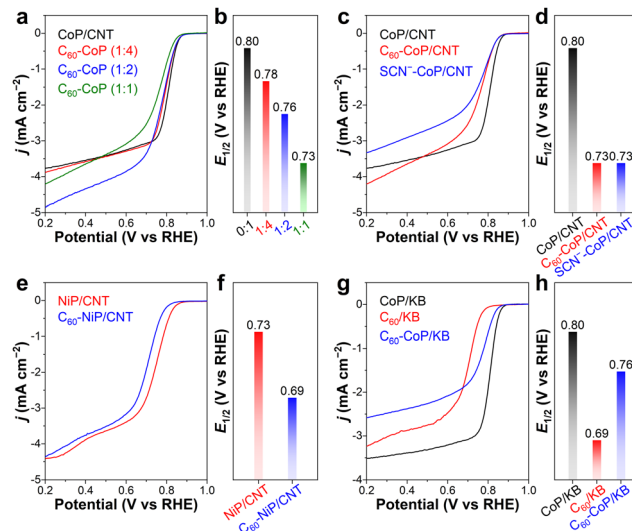
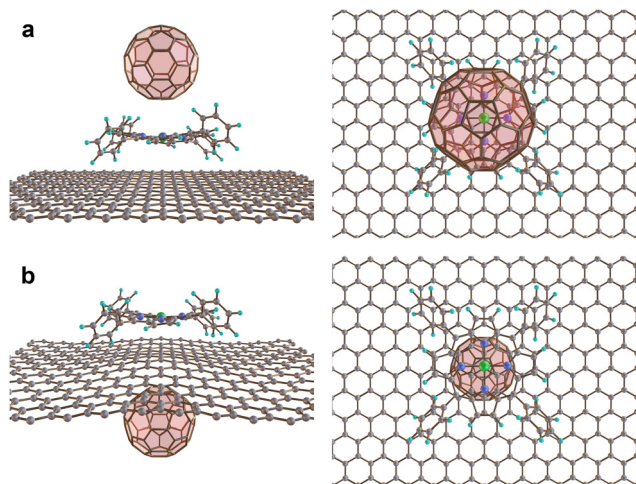


Fig. 3 (a) ORR polarization plots and (b) corresponding  $E_{1/2}$  of CoP/CNT and  $C_{60}$ -CoP/CNT with different  $C_{60}$ /CoP ratios of 1:4, 1:2, and 1:1. (c) ORR polarization plots and (d) corresponding  $E_{1/2}$  of CoP/CNT before and after the addition of  $C_{60}$  and  $SCN^-$ . (e) ORR polarization plots and (f) corresponding  $E_{1/2}$  of NiP/CNT and  $C_{60}$ -NiP/CNT. (g) ORR polarization plots and (h) corresponding  $E_{1/2}$  of CoP/KB,  $C_{60}$ /KB, and  $C_{60}$ -CoP/KB.

To rule out the potential influence of the exclusiveness of Co metal and the CNT support in the deactivation role of  $C_{60}$ , we have made further efforts by utilizing alternative MP molecules and a carbon support. Specifically, NiP/CNT and  $C_{60}$ -NiP/CNT are synthesized following the same synthetic recipe as that of the CoP counterparts. As displayed in Fig. 3e and f, substantial ORR activity loss is also observed for NiP/CNT after  $C_{60}$  adsorption with the  $E_{1/2}$  values reduced from 0.73 to 0.69 V, again manifesting the role of  $C_{60}$  as an inhibitor. Analogously, a conductive carbon black, ketjenblack (KB), is adopted to replace CNTs as the support of CoP molecules following the same synthetic methods. It is discerned that CoP/KB and  $C_{60}$ /KB exhibit almost the same ORR activities as those of CoP/CNT and  $C_{60}$ /CNT (Fig. 3g and h). After the addition of  $C_{60}$ , CoP/KB also experiences dramatic ORR activity decay with an evidently decreased  $E_{1/2}$  of 0.76 V. Therefore, it is confidently concluded that the probing role of fullerene will be universally applicable to various metal SACs and irrelevant to the carbon support.

Insights of the deactivation mechanism of  $C_{60}$  on CoP/CNT are further provided by DFT calculations. The preferable adsorption site of  $C_{60}$  on carbon-supported CoP is examined by calculating the binding energy (BE) of two distinct hybrids that have  $C_{60}$  and CoP on the same side (Fig. 4a) and opposite sides (Fig. 4b) of a graphene substrate, respectively. The results indicate that the hybrid with  $C_{60}$  adsorbed on top of CoP is more stable than the other by a  $\Delta BE$  value of 0.435 eV. Even when the solvent effect is included in the computation, a more stable model is obtained for  $C_{60}$  adsorbed to CoP (a  $\Delta BE$  value of 0.280 eV). Furthermore,  $O_2$  adsorption to the Co site has to overcome the considerable interactions between  $C_{60}$  and CoP, as reflected by the elongated distance between Co and  $C_{60}$  after



**Fig. 4** Geometric configurations of  $C_{60}$  adsorbed on the same side (a) and opposite sides (b) of CoP adsorbed on a graphene matrix. Views of the atomic models from two distinct angles are illustrated on the left and right, respectively. Same color code as in Fig. 2.

$O_2$  binding (Fig. S6, ESI<sup>†</sup>). Combined with the above experimental results, it is interpreted that the molecular SACs have probably been spatially blocked by the  $C_{60}$  molecules that are favorably adsorbed on top of the ORR-active Co atoms, creating substantial steric hindrance to restrict the access of  $O_2$ . Note that the inhibiting role of  $C_{60}$  to SACs does not contradict with the previously reported positive effects brought to metal particle catalysts by  $C_{60}$  adsorption.<sup>10–12</sup> In fact, it rules out the possibility of the surface metal atom, to which the  $C_{60}$  molecule is specifically adsorbed, to serve as the potential active sites.

In summary, utilizing metalloporphyrin molecules that are uniformly dispersed on CNTs *via* non-covalent  $\pi$ - $\pi$  interactions as model SACs, we disclose the probing role of  $C_{60}$  molecules in identifying the metal active sites of SACs towards the ORR. While the dispersion of CoP and  $C_{60}$  on CNTs leads to much superior ORR activities to bare CoP and  $C_{60}$ , respectively, the addition of  $C_{60}$  to CoP/CNT results in reduced ORR activity. A set of control experiments point out that the inhibiting effect of CoP gets severe with increasing  $C_{60}$  molar ratios.  $C_{60}$  leads to the same ORR activity decay as the recognized  $SCN^-$  poison. The probing role of  $C_{60}$  is also applicable to other metal SACs and carbon supports. Structural analysis and DFT calculations consistently indicate that the surface-stabilized CoP molecules are tightly confined between the  $C_{60}$  molecules and the basal plane of the CNTs. High steric hindrance is thus created to restrain the access of  $O_2$  to the central metal atoms of MP SACs.

This work was supported by the National Natural Science Foundation of China (22301088, 21925104, 92261204) and the Hubei Provincial Natural Science Foundation of China (2022CFB623, 2021CFA020). The authors also acknowledge the support from the Analytical and Testing Center in Huazhong University of Science and Technology.

## Data availability

The data supporting the findings of this study are available within the article and its ESI<sup>†</sup>.

## Conflicts of interest

There are no conflicts to declare.

## References

- 1 A. R. Puente Santiago, O. Fernandez-Delgado, A. Gomez, M. A. Ahsan and L. Echegoyen, *Angew. Chem., Int. Ed.*, 2021, **60**, 122–141.
- 2 K. Guo, N. Li, L. Bao and X. Lu, *Green Energy Environ.*, 2024, **9**, 7–27.
- 3 N. K. Gupta, A. Pashigeva, E. A. Pidko, E. J. Hensen and L. Mleczko, *et al.*, *Angew. Chem., Int. Ed.*, 2016, **55**, 1728–1732.
- 4 C. Zhang, W. Shen, K. Guo, M. Xiong and J. Zhang, *et al.*, *J. Am. Chem. Soc.*, 2023, **145**, 11589–11598.
- 5 N. Li, K. Guo, M. Li, X. Shao and Z. Du, *et al.*, *J. Am. Chem. Soc.*, 2023, **145**, 24580.
- 6 N. Li, M. Li, K. Guo, Z. Guo and R. Wang, *et al.*, *Adv. Energy Mater.*, 2024, **14**, 2401008.
- 7 M. F. Sanad, H. M. Franklin, B. A. Ali, A. R. Puente Santiago and A. N. Nair, *et al.*, *Angew. Chem., Int. Ed.*, 2022, **61**, e202116727.
- 8 T. Wang, L. Zhang, J. Wu, M. Chen and S. Yang, *et al.*, *Angew. Chem., Int. Ed.*, 2023, **62**, e202311352.
- 9 B. Peng, *J. Am. Chem. Soc.*, 2022, **144**, 19921–19931.
- 10 J. Zheng, L. Huang, C. H. Cui, Z. C. Chen and X. F. Liu, *et al.*, *Science*, 2022, **376**, 288–292.
- 11 B. Zhao, F. Chen, C. Cheng, L. Li and C. Liu, *et al.*, *Adv. Energy Mater.*, 2023, **13**, 2204346.
- 12 Y. Li, H. Zhang, T. Chen, Y. Sun and F. Rosei, *et al.*, *Adv. Funct. Mater.*, 2024, **34**, 2312970.
- 13 Y. Feng, X. Wang, J. Huang, P. Dong and J. Ji, *et al.*, *Chem. Eng. J.*, 2020, **390**, 124525.
- 14 Y. H. Choi, J. Lee, A. Parija, J. Cho and S. V. Verkhoturov, *et al.*, *ACS Catal.*, 2016, **6**, 6246–6254.
- 15 A. R. Puente Santiago, T. He, O. Eraso, M. A. Ahsan and A. N. Nair, *et al.*, *J. Am. Chem. Soc.*, 2020, **142**, 17923–17927.
- 16 A. Zitolo, V. Goellner, V. Armel, M. T. Sougrati and T. Mineva, *et al.*, *Nat. Mater.*, 2015, **14**, 937–942.
- 17 F. Luo, A. Roy, L. Silvioli, D. A. Cullen and A. Zitolo, *et al.*, *Nat. Mater.*, 2020, **19**, 1215–1223.
- 18 K. M. Zhao, S. Liu, Y. Y. Li, X. Wei and G. Ye, *et al.*, *Adv. Energy Mater.*, 2022, **12**, 2103588.
- 19 X. Yang, D. Xia, Y. Kang, H. Du and F. Kang, *et al.*, *Adv. Sci.*, 2020, **7**, 2000176.
- 20 P. Hutchison, P. S. Rice, R. E. Warburton, S. Raugei and S. Hammes-Schiffer, *J. Am. Chem. Soc.*, 2022, **144**, 16524–16534.
- 21 T. Zhang, J. Jin, J. Chen, Y. Fang and X. Han, *et al.*, *Nat. Commun.*, 2022, **13**, 6875.
- 22 M. L. Pegis, C. F. Wise, D. J. Martin and J. M. Mayer, *Chem. Rev.*, 2018, **118**, 2340–2391.
- 23 J. Wu and Y. X. Yu, *J. Colloid Interface Sci.*, 2022, **623**, 432–444.
- 24 J. Wu and Y. X. Yu, *Int. J. Hydrogen Energy*, 2023, **48**, 5961–5975.
- 25 C.-X. Zhao, B.-Q. Li, J.-N. Liu, J.-Q. Huang and Q. Zhang, *Chin. Chem. Lett.*, 2019, **30**, 911–914.
- 26 E. M. Perez and N. Martin, *Chem. Soc. Rev.*, 2015, **44**, 6425–6433.
- 27 A. J. Néel, M. J. Hilton, M. S. Sigman and F. D. Toste, *Nature*, 2017, **543**, 637–646.
- 28 R. Gao, Q. Dai, F. Du, D. Yan and L. Dai, *J. Am. Chem. Soc.*, 2019, **141**, 11658–11666.
- 29 S. Gupta, C. Fierro and E. Yeager, *J. Electroanal. Chem. Interfacial Electrochem.*, 1991, **306**, 239–250.
- 30 C. H. Bartholomew, *Appl. Catal., A*, 2001, **212**, 17–60.
- 31 M. S. Thorum, J. M. Hankett and A. A. Gewirth, *J. Phys. Chem. Lett.*, 2011, **2**, 295–298.
- 32 Q. Wang, Z. Y. Zhou, Y. J. Lai, Y. You and J. G. Liu, *et al.*, *J. Am. Chem. Soc.*, 2014, **136**, 10882–10885.