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Enhanced H₂ production assisted by anodic iodide oxidation using transparent tin oxide-based electrodes†

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In this work, the direct use of transparent conducting oxides (TCOs) as cost-efficient anodes for the iodide oxidation reaction (IOR) is explored. Energy-saving hydrogen production assisted by the IOR is demonstrated using a hybrid water electrolysis system with FTO as the anode and Pt-wire as the cathode. The hybrid system delivers 10 mA cm⁻² at a cell voltage as low as 1.15 V with the faradaic efficiency for H₂ found to be ~91%. This study may open avenues for developing novel systems that integrate the IOR with other high-value reduction reactions.

Hydrogen (H₂)-based energy systems are promising for producing clean and sustainable energy owing to its high gravimetric heating value (141.9 MJ kg⁻¹) and zero emissions on combustion.^{1,2} H₂ production can be sustainable if generated through water electrolysis using renewable energy sources.¹ However, the anodic oxygen evolution reaction (OER) is a bottleneck in water splitting owing to inherently high energy demand due to its multi-step multi-electron nature, which is known to limit the efficiency of water electrolysis.³ If the OER is replaced with thermodynamically more readily oxidizable molecules (whose oxidation potential is lower than the OER) such as alcohols,⁴ hydrazine,⁵ urea,⁶ benzylamine,⁷ hydroxymethyl-furfural (HMF),⁸ etc., then energy-saving H₂ production can be realized.⁹ These hybrid water electrolysis systems are capable of generating H₂ and high-value products concomitantly.⁹ The iodide oxidation reaction (IOR) to iodine (I₂) is one such reaction whose thermodynamic potential is 0.54 V vs. NHE in acidic medium,^{10a} which is ~700 mV lower than that of the OER, and thus may lead to facile H₂ production when hydrogen evolution reaction (HER) is integrated with the IOR.^{10a} Recently, the electrocatalytic IOR has attracted significant attention for achieving energy-saving H₂ production.¹⁰ In addition, the product of the IOR, for instance, I₂ is a high-value chemical and widely used in the hygiene industry, pharmaceuticals, chemical synthesis, etc.^{10e,11} in comparison to less

economic oxygen which is the product of the OER in traditional water electrolysis. Earlier studies on IOR have reported the use of materials which are either not cost-efficient or require stringent experimental conditions.^{10a,b,d,f} The continuous quest for the development of novel and cost-efficient electrocatalysts prompted researchers to alleviate the dependency on noble-metal based materials in energy conversion devices. Transparent conducting oxide (TCO)-based materials, namely fluorine doped tin oxide (FTO), indium-tin oxide (ITO), and antimony-doped tin oxide (ATO), have been extensively used for various applications.¹² TCOs are typically degenerately-doped wide band gap (3 eV) semiconductors possessing fascinating properties, such as high optical transparency (>80% at 550 nm for FTO) in the visible region of the electromagnetic spectrum, low electrical resistivity (~10⁻⁴ Ω cm for FTO), etc.^{12,13} The majority of the studies use TCOs as conductive supports. For instance, FTO based electrodes have been used as transparent conductive substrates in solar cells, and organic light emitting diodes (OLEDs).¹² Additionally, the TCOs find their existence in polymer electrolyte membrane (PEM) fuel cells, however as a conductive support for electrocatalysts.¹⁴ In other words, modified FTO/ITO based TCOs have been extensively studied for (photo)electrochemical applications. However, it is noteworthy that the direct use of TCOs as an electrocatalyst is scarce.^{15,16} The direct use of TCOs may attract attention due to the possibility of realizing practical applications, considering the commercial availability of large area FTO electrodes. Additionally, the direct use of transparent FTO provides scope to develop photoelectrochemical IOR systems that can be integrated with the HER or CO₂ reduction reaction (CO₂RR). Herein, the direct use of TCOs, namely FTO and ITO, in pristine form as an electrocatalyst for the IOR is studied. In addition to the high conductivity offered by TCOs, the appreciable stability of TCOs, especially of FTO in acidic environments,^{13,17} led us to choose TCOs in the present study. The ability of FTO for the IOR is studied and installed the same as anodes for the first time in hybrid water electrolysis that couples the HER and IOR.

The as-obtained commercial TCOs such as FTO and ITO coated glass substrates after thorough cleaning were used as electrodes in the present study. The basic characterization studies were carried

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out to ensure the characteristics of the FTO and ITO electrodes. Fig. S1 (ESI†) shows the XRD pattern of FTO, which matches well with the *cassiterite* phase of SnO_2 (JCPDS file no. #00-41-1445). The XRD pattern of ITO consisted of reflections corresponding to SnO_2 (JCPDS file no. #00-41-1445) and In_2O_3 (JCPDS file no. #00-006-0416), as expected.¹⁸ The Raman spectra of FTO and ITO show characteristic Raman bands at 115, 435, 556, and 772 cm^{-1} which are assigned for the B_{1g} , E_g , E_u and B_{2g} modes, respectively for *cassiterite* SnO_2 (Fig. S2, ESI†).¹⁹ Fig. S3 (ESI†) shows the morphology of the FTO and ITO coated glass substrates obtained from field emission scanning electron microscopy (FESEM) images.

Electrochemical activity of FTO and ITO coated glass substrates were studied in Ar-saturated 0.1 M HClO_4 at 25°C in a standard three-electrode system. Pre-cleaned FTO/ITO coated glass was directly used as the working electrode (WE), and a saturated calomel electrode (SCE) and graphite rod acted as the reference electrode (RE) and counter electrode (CE), respectively. Fig. 1a represents the *iR*-corrected (80%) linear sweep voltammograms (LSVs) depicting OER and IOR activity when the measurements were carried out in the absence and presence of NaI, respectively. It is noted that the potential required to observe OER currents is very high indicating that FTO is not an efficient OER electrode. However, in the presence of NaI, the voltammograms show a clear and dominant oxidation of iodide at low potentials, signifying the fact that FTO can oxidize I^- to I_2 . It is observed that the FTO electrode requires a potential of $\sim 1.04\text{ V}$ vs. RHE to reach the benchmark current density of 10 mA cm^{-2} for the IOR. Table S1 (ESI†) presents the comparison of the performance of FTO for the IOR with some of the recently reported electrocatalysts. Furthermore, it is observed that the currents increase with increase in the concentration of NaI, confirming that the currents originated in the presence of NaI are due to IOR at the FTO surface (Fig. 1b). The LSV measurements with lower concentrations of I^- shown in Fig. S4 (ESI†) indicate that FTO is capable of oxidizing even low concentrations of I^- . It should be noted that by

replacing the OER by IOR, the potential required to reach 1 mA cm^{-2} can be significantly lowered by $\sim 1.65\text{ V}$. The data clearly suggest that the IOR can be a potential alternative to the OER to realize energy-saving H_2 production. Furthermore, when the electrode is pre-treated at 3.3 V vs. RHE (outside the stability limits of FTO)^{13,17} for 10 min prior to IOR measurements, the IOR activity substantially decreased as the chosen potential is known to leach out some of the Sn-ions from the FTO.¹³ As shown in Fig. S5 (ESI†), the potential required to reach 10 mA cm^{-2} is increased from 1.09 V vs. RHE for pristine FTO to 1.51 V vs. RHE for pre-treated FTO, suggesting the role of Sn-ions in dictating IOR activity. Additionally, the ability of FTO towards IOR activity is further tested by carrying out measurements in alkaline (0.1 M NaOH) solutions of NaI (Fig. 1c), as the product of the IOR depends on pH.^{10b} For instance, the product of the IOR at low pH is predominantly iodine ($2\text{I}^- \rightarrow \text{I}_2 + 2\text{e}^-$; $\text{pH} \sim 1$),^{10b} while it is iodate when the IOR is performed in high pH ($\text{I}^- + 6\text{OH}^- \rightarrow \text{IO}_3^- + 3\text{H}_2\text{O} + 6\text{e}^-$; $\text{pH} \sim 13$)^{10b} conditions. As indicated with digital pictures in Fig. 1c, the generation of yellow colour product formed at low pH conditions (in HClO_4 medium) confirms the formation of iodine (in the form of I_3^-), while it is colourless at high pH conditions (in NaOH medium) as the product formed is IO_3^- , which is colourless. It should be noted that the IOR in alkaline medium requires higher potentials than that in acidic medium due to the complex 6-electron process.^{10b,11} Therefore, further studies in the present study are not carried out in an alkaline medium. The IOR studies performed using ITO coated glass are given in Fig. S6 (ESI†). The data indicate that FTO and ITO show comparable IOR activity, with ITO exhibiting relatively higher current densities as compared to FTO. Electrochemical impedance spectroscopy (EIS) data for the OER and IOR recorded using FTO and ITO are given in Fig. S6(b) (ESI†). Among FTO & ITO, the radius of the semicircle for the IOR is noted to be smaller for ITO indicating the lower resistance for charge transfer at the electrode/electrolyte interface than that of FTO. The Nyquist plot of the OER for both FTO and ITO suggests a higher charge transfer resistance than IOR, reconfirming the facile oxidation of iodide (0.54 V vs. NHE) than water (1.23 V vs. NHE) on FTO and ITO electrodes. Though ITO performed slightly better than FTO for the IOR, further studies were carried out with FTO due to its relatively higher stability than ITO in acidic environments.^{13,17} The stability aspects will be discussed in detail in the subsequent section.

To gain insight into the reaction kinetics at the FTO/electrolyte interface and the electron transfer, *in situ* EIS measurements were performed in 0.1 M HClO_4 with and without 1 M NaI at different DC potentials (Fig. S7a and b, ESI†). As can be seen from the Bode phase plots of the OER in Fig. S7a (ESI†), FTO showed no frequency peak between 1.65 V and 1.9 V vs. RHE in 0.1 M HClO_4 without NaI. The commencement of a peak in the low-frequency range started at $\sim 1.95\text{ V}$ vs. RHE, which can be attributed to the adsorption of oxygen active species on the surface and the corresponding faradaic reaction (OER),²⁰ complementing the LSV results. After the addition of 1 M NaI (Fig. S7b, ESI†), the peak in the low-frequency range started to appear from 0.65 V vs. RHE indicating that iodide ions adsorb on the FTO surface in a more favourable manner. This accounts for the lowering of the potential by $\sim 1300\text{ mV}$.

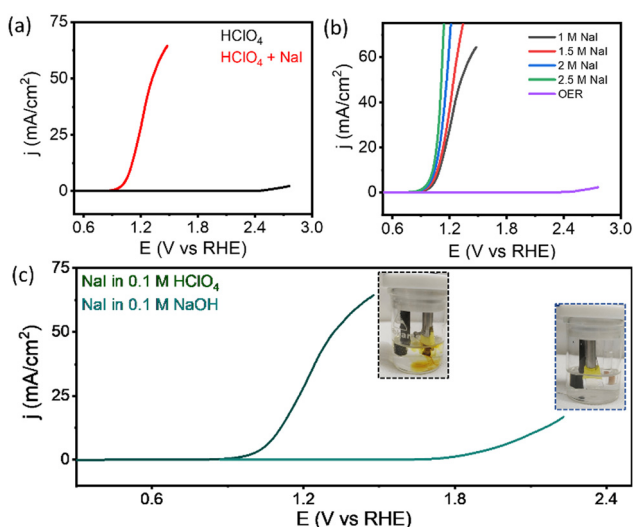


Fig. 1 *iR*-corrected (a) linear sweep voltammograms obtained using FTO with (red) and without (black) 1 M NaI in 0.1 M HClO_4 , (b) variation of the voltammograms with concentration of NaI, (c) voltammograms recorded when 1 M NaI dissolved in different electrolytes. The scan rate used was 5 mV s^{-1} .



Motivated by the preliminary electrochemical measurements, the product quantification studies were performed using an H-type two-compartment cell (Fig. S8, ESI†). A commercial Pt-wire was chosen as the CE due to its high HER activity in acidic electrolytes. The use of a divided cell eliminates the unwanted counter reactions to the IOR (reduction of product formed in the IOR) at the CE. The H-cell setup was tested by recording voltammograms prior to quantification studies. The data obtained in the H-type cell (Fig. 2a) are similar to the LSVs recorded in the 3-electrode cell given in Fig. 1a. The products formed at the CE (H_2) and WE (I_2) when the system was held at a potential of 1.08 V vs. RHE were quantified using gas chromatography (GC) and UV-visible spectroscopy, respectively. Fig. 2b represents typical GC traces recorded for the gaseous products collected from the CE compartment, confirming the formation of H_2 . Similarly, Fig. 2c shows typical absorption spectra of the liquid product (with a dilution factor of 8 \times) collected from the WE compartment at different intervals of time. The data clearly show the appearance of a peak corresponding to I_3^- (at 350 nm) that grows with time, confirming that the oxidation of I^- leads to the formation of I_2 (since I_2 in the presence of I^- , exists as I_3^-). The concentration of iodine in the form of I_3^- was estimated using a calibration plot obtained by recording absorption spectra of various concentrations of standard iodine (dissolved in I^-) (Fig. S9 (ESI†)). The faradaic efficiency (FE) is found to be $\sim 95\%$ for the gaseous product (H_2) while the FE value for the liquid product ($I_2(I_3^-)$) is $\sim 98\%$ (Fig. 2d). To confirm the formation of I_2 during the IOR process, the electrolyte obtained after long-term chronoamperometry tests carried out at 1.1 V vs. RHE for 14 h, was collected and subjected to extraction of I_2 in chloroform, as shown in Fig. S10a (ESI†). The absorption spectrum of the extract (Fig. S10b, ESI†) shows a peak at 511 nm which is similar to that

of the standard I_2 , confirming the formation of I_2 as the product of the IOR in acidic medium. Furthermore, the robustness of the FTO electrode is evaluated by performing long-term (24 h) tests at 10 mA cm^{-2} . As shown in Fig. S11 (ESI†), the FTO electrode exhibited stable electrochemical performance. The FTO electrode after long term testing was characterised using XRD, XPS and Raman spectroscopy to decipher the stability of the electrode. The XRD patterns (Fig. S12, ESI†) and Raman spectra (Fig. S13, ESI†) of FTO before and after the long term stability test show negligible changes indicating the robustness of the FTO substrate. Additionally, the surface morphology of the FTO electrodes remained intact after the long-term test as evidenced by FESEM images shown in Fig. S14 (ESI†). Notably, the energy dispersive X-ray spectrum (EDS) clearly shows the presence of I-species suggesting the adsorption of I-species on the FTO surface. The XPS (Fig. S15, ESI†) data of the FTO electrode after the stability test show characteristic peaks corresponding to Sn 3d and O 1s that do not show any significant variation. The presence of I-species is noted from the I 3d XPS data of the post catalysis sample (Fig. S15d, ESI†), complementing the EDS data. As mentioned above, the characterization studies of the post catalysis ITO electrode reveal relatively poorer stability than FTO (More details in ESI† Fig. S16). It should be noted that spontaneous oxidation of I^- under acidic conditions ($4H^+ + 6I^- + O_2 \rightarrow 2H_2O + 2I_3^-$) can be a competitive reaction which may generate iodine (in the form of I_3^-). To understand this, a control experiment was carried out at 25 $^{\circ}C$, similar to the conditions used in electrochemical experiments; however in the absence of applied voltage. As shown in Fig. 2e, the formation of iodine or I_3^- is not very evident in voltage-free conditions as compared to that obtained *via* electrochemical oxidation carried out at 1.17 V vs. RHE. It is observed that the amount of triiodide produced by electrochemical oxidation is 63.4 mM while it is 23 μM in control experiments after 22 h (see ESI† Fig. S17 and S18). This clearly suggests the electrocatalytic effect of the FTO electrode in oxidizing iodide to iodine. Additionally, the IOR in a neutral medium (0.1 M Na_2SO_4) was also studied and the product formed during IOR was quantified (see Fig. S19, ESI†).

Further, hybrid water electrolysis (IOR||HER) measurements were performed in an H-type electrochemical cell as mentioned above, using optimized NaI concentrations (Fig. S20, ESI†) with FTO as the anode and Pt-wire as cathode (FTO||Pt-wire) (Fig. 3a). The polarization curves obtained at a scan rate of 5 $mV s^{-1}$ are shown in Fig. 3b. The data suggest that to achieve a current density of 10 mA cm^{-2} , a cell voltage of 1.15 V is required in hybrid water electrolysis, which is comparable to several hybrid water electrolysis systems recently reported in the literature (Table S2, ESI†). Additionally, the energy-saving efficiency at 1 mA cm^{-2} is found to be 56% when the IOR is integrated with the HER. Furthermore, the H_2 generated in the hybrid electrolyzer was also quantified by holding the cell at a cell voltage of 1.15 V. A FE of around 91% was observed for H_2 . It should be noted that H_2 production is not observed in traditional water electrolysis setups under the conditions (1.15 V) used in hybrid water electrolysis (Fig. S21, ESI†). Fig. 3c shows the long-term stability of FTO in the two electrode-electrolyzer setup when the system is at 10 mA cm^{-2} and the cell voltage-time data

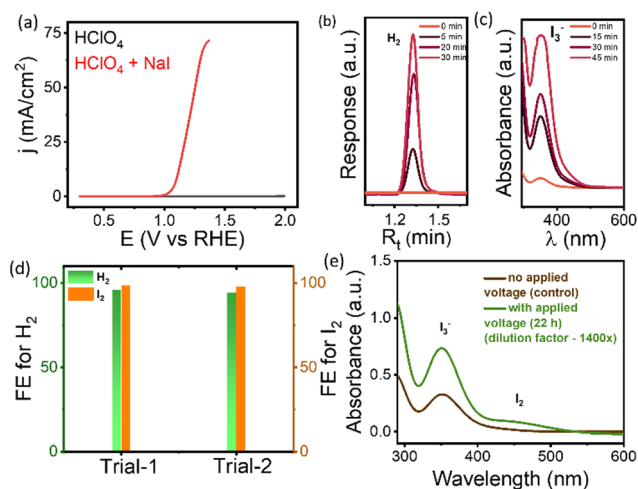


Fig. 2 (a) *iR*-corrected linear sweep voltammetry curves obtained in a H-cell using FTO as WE and Pt-wire as CE in the presence and absence of 1 M NaI in 0.1 M $HClO_4$, at a scan rate of 5 $mV s^{-1}$, (b) and (c) representative gas chromatograms and UV-Vis absorbance spectra, respectively, recorded during the course of potentiostatic measurements at 1.08 V vs. RHE, (d) faradaic efficiency of products H_2 and I_2 determined in two different trials, and (e) UV-vis absorbance spectra of the aliquot of the product obtained with and without applied voltage.

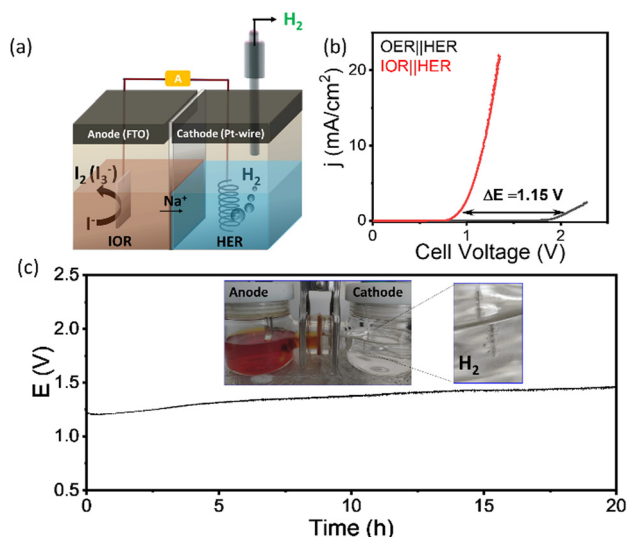


Fig. 3 (a) Schematic illustration of a hybrid water electrolysis cell consisting of Pt-wire as the cathode and FTO as the anode, (b) linear sweep voltammetry curves obtained for hybrid water electrolysis (red) and traditional water electrolysis (black), (c) variation of the cell voltage with time monitored when the hybrid water electrolysis system is held at 10 mA cm⁻². Inset of (c) represents the snapshot depicting the formation of iodine at the anode and H₂ at the cathode captured during the electrolysis.

suggest that even after 20 h, the FTO shows only a little change in the cell voltage of 12.87 mV h⁻¹ as compared to the initial value, reaffirming the stability of FTO for the IOR. It is worth noting that the pristine FTO as an anode for the IOR reported in the present study is capable of delivering efficiencies comparable to several of the reported systems that use anodes either noble metal-based or that require tedious experimental conditions. This study may open up new avenues for the further exploration of the use of FTO-based electrodes for energy-saving H₂ production assisted by an IOR process.

In summary, the use of TCOs, especially FTO, as an efficient electrode for the IOR is demonstrated. The IOR is proven to be a potential alternative to the OER. Impressively, FTO electrodes exhibited high activity for the IOR with a FE value of 98%. The integration of the IOR with the HER in hybrid water electrolysis enables the realization of efficient H₂ production at less input voltage, when FTO is used as the anode and Pt is used as the cathode. The hybrid water electrolysis cell (FTO||Pt-wire) delivered 10 mA cm⁻² at a cell voltage of only 1.15 V with 91% FE for H₂ production.

SP has carried out the experimental work and the data analysis. KV conceived the idea, procured funding, and monitored the execution and data analysis of the project. Shivam Nikam, undergraduate student is acknowledged for initial assistance. The manuscript was written through the contributions of all authors. All authors have given approval to the final version of the manuscript.

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Data availability

Data are available upon request to the corresponding author.

Conflicts of interest

There are no conflicts to declare.

Notes and references

- 1 I. Staffell, D. Scamman, A. Velazquez Abad, P. Balcombe, P. E. Dodds, P. Ekins, N. Shah and K. R. Ward, *Energy Environ. Sci.*, 2019, **12**, 463–491.
- 2 (a) J. H. Kim, D. Hansora, P. Sharma, J. W. Jang and J. S. Lee, *Chem. Soc. Rev.*, 2019, **48**, 1908–1971; (b) L. Quan, X. Chen, J. Liu, S. Fan, B. Y. Xia and B. You, *Adv. Funct. Mater.*, 2023, **33**, 2307643; (c) F. Song, T. Zhang, D. Zhou, P. Sun, Z. Lu, H. Bian, J. Dang, H. Gao, Y. Qian, W. Li, N. Jiang, H. Dummer, J. G. Shaw, S. Chen, G. Chen, Y. Sun and Y. Rao, *ACS Mater. Lett.*, 2022, **4**, 967–977.
- 3 B. You and Y. Sun, *Acc. Chem. Res.*, 2018, **51**, 1571–1580.
- 4 J. Li, L. Li, X. Ma, X. Han, C. Xing, X. Qi, R. He, J. Arbiol, H. Pan, J. Zhao, J. Deng, Y. Zhang, Y. Yang and A. Cabot, *Adv. Sci.*, 2023, **10**, 2300841.
- 5 (a) L. Zhu, J. Huang, G. Meng, T. Wu, C. Chen, H. Tian, Y. Chen, F. Kong, Z. Chang, X. Cui and J. Shi, *Nat. Commun.*, 2023, **14**, 1997; (b) S. Ma, B. Yu, B. Y. Xia and B. You, *Sci. China Mater.*, 2024, **67**, 752–761.
- 6 Y. Wang, L. Chen, H. Zhang, M. Humayun, J. Duan, X. Xu, Y. Fu, M. Boudoudina and C. Wang, *Green Chem.*, 2023, **25**, 8181–8195.
- 7 I. Mondal, P. V. Menezes, K. Laun, T. Diemant, M. Al-Shakran, I. Zebger, T. Jacob, M. Driess and P. W. Menezes, *ACS Nano*, 2023, **17**, 14043–14052.
- 8 T. Wei, W. Liu, S. Zhang, Q. Liu, J. Luo and X. Liu, *Chem. Commun.*, 2023, **59**, 442–445.
- 9 (a) E. A. Moges, C.-Y. Chang, M.-C. Tsai, W.-N. Su and B. J. Hwang, *EES Catal.*, 2023, **1**, 413–433; (b) G. Chen, X. Li and X. Feng, *Angew. Chem., Int. Ed.*, 2022, **61**, e202209014.
- 10 (a) D. B. Adam, M. C. Tsai, Y. A. Awoke, W. H. Huang, Y. W. Yang, C. W. Pao, W. N. Su and B. J. Hwang, *ACS Sustainable Chem. Eng.*, 2021, **9**, 8803–8812; (b) D. B. Adam, M. C. Tsai, Y. A. Awoke, W. H. Huang, C. H. Lin, T. Alamirew, A. A. Ayele, Y. W. Yang, C. W. Pao, W. N. Su and B. J. Hwang, *App. Catal., B.*, 2022, **316**, 121608; (c) E. Hu, Y. Yao, Y. Chen, Y. Cui, Z. Wang and G. Qian, *Nanoscale Adv.*, 2021, **3**, 604–610; (d) Y. S. Park, G. Jang, I. Sohn, H. Lee, J. Tan, J. Yun, S. Ma, J. Lee, C. U. Lee, S. Moon, H. Im, S. M. Chung, S. Yu, H. Kim and J. Moon, *Carbon Energy*, 2023, **5**, e366; (e) S.-M. Peng, S. B. Patil, C.-C. Chang, S.-T. Chang, Y.-C. Chen, K.-C. Wu, W.-N. Su, B. J. Hwang and D.-Y. Wang, *J. Mater. Chem. A*, 2022, **10**, 23982–23989; (f) T. A. Dessie, W. H. Huang, D. B. Adam, Y. A. Awoke, C. H. Wang, J. L. Chen, C. W. Pao, N. G. Habtu, M. C. Tsai, W. N. Su and B. J. Hwang, *Nano Lett.*, 2022, **22**, 7311–7317; (g) S. Paniya, A. D. Gaonkar and K. Vankayala, *Chem. Commun.*, 2024, **60**, 4174–4177.
- 11 L. Yao, Y. P. Liu, H. H. Cho, M. Xia, A. Sekar, B. P. Darwich, R. A. Wells, J. H. Yum, D. Ren, M. Gratzel, N. Guijarro and K. Sivula, *Energy Environ. Sci.*, 2021, **14**, 3141–3151.
- 12 P. P. Edwards, A. Porch, M. O. Jones, D. V. Morgan and R. M. Perks, *Dalton Trans.*, 2004, 2995–3002.
- 13 S. Geiger, O. Kasian, A. M. Mingers, K. J. J. Mayrhofer and S. Cherevko, *Sci. Rep.*, 2017, **7**, 4595.
- 14 J. M. Kim, Y. J. Lee, S. Kim, K. H. Chae, K. R. Yoon, K. A. Lee, A. Byeon, Y. S. Kang, H. Y. Park, M. K. Cho, H. C. Ham and J. Y. Kim, *Nano Energy*, 2019, **65**, 104008.
- 15 A. Herman, J. L. Mathias and R. Neumann, *ACS Catal.*, 2022, **12**, 4149–4155.
- 16 G. Gobel, A. Talke and F. Lisdat, *Electroanalysis*, 2018, **30**, 225–229.
- 17 J. D. Benck, B. A. Pinaud, Y. Gorlin and T. F. Jaramillo, *PLoS One*, 2014, **9**, e107942.
- 18 L. Cojocar, C. Olivier, T. Toupance, E. Sellier and L. Hirsch, *J. Mater. Chem. A*, 2013, **1**, 13789.
- 19 A. Korjenic and K. S. Raja, *J. Electrochem. Soc.*, 2019, **166**, C169–C184.
- 20 S. Bai, L. Chen, J. Bai, C. Lv, S. Xu, D. Zhang, H. Meng, C. Guo, H. Yang and C. Shang, *Inorg. Chem. Front.*, 2023, **10**, 4695–4701.

