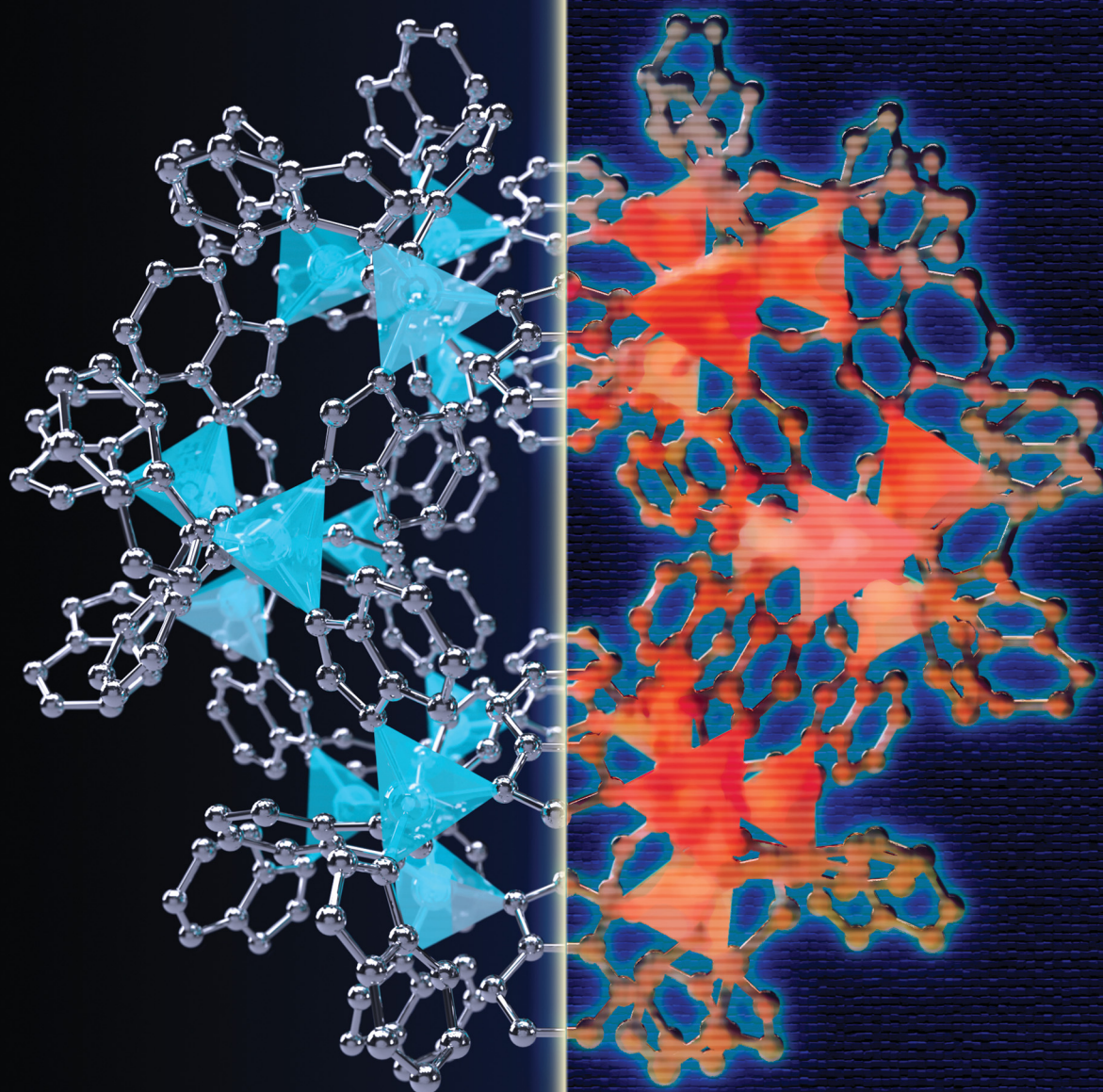


ChemComm

Chemical Communications

rsc.li/chemcomm



ISSN 1359-7345

COMMUNICATION

Masataka Ohtani *et al.*
Thermodynamic analysis of gate-opening carbon dioxide
adsorption behavior of metal-organic frameworks



Cite this: *Chem. Commun.*, 2024, 60, 4170

Received 21st November 2023,
Accepted 26th February 2024

DOI: 10.1039/d3cc05700c

rsc.li/chemcomm

Thermodynamic analysis of gate-opening carbon dioxide adsorption behavior of metal–organic frameworks†

Shino Kannaka, Ayumi Ohmiya, Chiho Ozaki and Masataka Ohtani *

Thermodynamic analysis of gate-opening carbon dioxide (CO₂) adsorption behavior of metal–organic frameworks (MOFs) was investigated using differential scanning calorimetry (DSC). Unlike measurements under nitrogen atmosphere, obvious exothermic and endothermic peaks were observed in DSC curves under CO₂ flow. *In situ* heating X-ray diffraction and thermogravimetric analyses under CO₂ revealed that reversible crystal structure and weight changes occurred upon CO₂ adsorption/desorption. The thermodynamic parameters of the CO₂ adsorption process by MOFs were determined by DSC analysis at various CO₂ partial pressures.

Metal–organic frameworks (MOFs), which consist of alternating coordination bonds of metal ions and organic ligands, can be designed with pore structures. MOFs with pore sizes and shapes that are appropriate for different applications can be produced by changing the combination of metal ions and organic ligands.^{1,2} However, the robust structure of most MOFs makes it difficult to control the movement of gas molecules in the pores during gas adsorption and separate mixed gases efficiently. Fortunately, some MOFs are known to change their structure in response to certain stimuli. For example, flexible MOFs with crystalline structures that open up at a certain atmospheric pressure and abruptly adsorb gas molecules have been reported.^{3–6} As the pressure decreases, the MOF structures revert to their closed forms, releasing the gas molecules. This gas adsorption behavior, driven by crystal structure changes, is called gate-opening adsorption. The gas adsorption/desorption isotherms of gate-opening MOFs exhibit stepwise adsorption/desorption curves, which cannot be classified by the IUPAC classification for common porous materials.⁷ Such behavior can be used for the selective separation of mixed gases because different gas molecules have different threshold pressures for the gate opening of the MOF structure.

Zeolitic imidazole framework-7 (ZIF-7) is a well-known MOF that shows structural changes upon adsorption and desorption of carbon dioxide (CO₂).^{8–10} ZIF-7 is composed of zinc(II) ions (Zn²⁺) and benzimidazole (BIm) and has three known crystal polymorphs (phases I, II, and III). These polymorphs have different coordination network structures because of their different bond angles of BIm with Zn²⁺. Phases I, II, and III of ZIF-7 have rhombohedral, triclinic, and monoclinic crystal systems, respectively. The structure of ZIF-7 changes to phase II upon desorption of *N,N*-dimethylformamide (DMF) from the pores of the phase-I polymorph. Phase I is recovered from phase II by the adsorption of DMF in the closed pores of phase II. These reversible conformational changes of ZIF-7 between phases I and II can also be induced by CO₂ adsorption/desorption.^{11,12}

In CO₂ adsorption measurements, ZIF-7 shows a staircase adsorption isotherm as the CO₂ gas pressure changes, which is characteristic of gate-opening MOFs.^{13–16} The structural change of ZIF-7 induced by CO₂ adsorption has been examined by *in situ* heating powder X-ray diffraction (XRD).^{16,17} The CO₂ adsorption process of ZIF-7 has also been measured under various pressures using synchrotron powder XRD.¹⁸ Kamili *et al.*¹⁹ traced structural changes of ZIF-7 with temperature and pressure using Raman spectroscopy. Molecular dynamics simulations have predicted how CO₂ moves through adsorption sites of ZIF-7.^{17,18} In addition, neutron powder diffraction has been used to identify preferential sites of ZIF-7 for CO₂ adsorption.^{18,20} Density functional theory (DFT) was used to estimate the thermodynamic energy change required for CO₂ adsorption at each location.¹⁸

Although the above studies have investigated the relationship between CO₂ adsorption and changes in ZIF-7 crystal structure, this relationship still has many unknown parameters. In particular, the interplay between the CO₂ adsorption process, changes in ZIF-7 structure, and adsorption energy remains unclear. Further detailed experimental investigation of the CO₂ adsorption process by ZIF-7 under atmospheric pressure should be conducted. In this study, differential scanning calorimetry (DSC) is performed under CO₂ atmosphere to evaluate the calorimetric changes accompanying the gate-opening structural

School of Engineering Science, Kochi University of Technology, 185 Miyanokuchi, Tosayamada, Kami, Kochi 782-8502, Japan.

E-mail: ohtani.masataka@kochi-tech.ac.jp

† Electronic supplementary information (ESI) available. See DOI: <https://doi.org/10.1039/d3cc05700c>



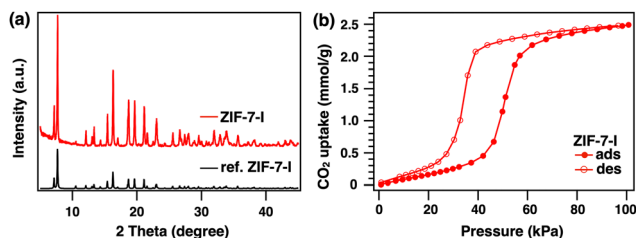


Fig. 1 (a) XRD pattern of as-prepared ZIF-7-I (red) and simulated reference pattern of ZIF-7-I (black, CCDC 602541). (b) CO₂ adsorption/desorption isotherms of ZIF-7-I at 27 °C.

changes of ZIF-7 during CO₂ adsorption/desorption. The behavior of ZIF-7 during CO₂ adsorption/desorption under various conditions is analyzed by DSC to thermodynamically evaluate the adsorption mechanism. Changes in adsorption/desorption behavior associated with structural changes are also followed by *in situ* heating powder XRD and thermogravimetric analysis (TGA) measurements. Furthermore, DSC analysis under various CO₂ partial pressures ($p(\text{CO}_2)$) is used to evaluate the thermodynamic parameters of CO₂ adsorption for ZIF-7.

First, the synthetic conditions for ZIF-7 were thoroughly investigated by modifying the literature methods (see the ESI† for details).¹² Fig. 1a shows powder XRD patterns of the ZIF-7 product (denoted hereafter as ZIF-7-I, where I indicates phase I). The observed diffraction peaks were matched to reference peaks, confirming that highly crystalline ZIF-7-I was obtained. Next, CO₂ adsorption measurements were performed to determine the CO₂ adsorption capacity of ZIF-7-I. Prior to the adsorption measurements, DMF in the pores of ZIF-7-I was removed by heating at 200 °C for 20 h. TGA results confirmed the removal of DMF from ZIF-7-I (Fig. S1, ESI†). CO₂ adsorption/desorption measurements were performed at 27 °C. The obtained ZIF-7-I showed CO₂ adsorption/desorption isotherms consistent with gate-opening behavior (Fig. 1b), which rose sharply at a certain pressure. The maximum CO₂ uptake of ZIF-7-I was 2.5 mmol g⁻¹, which was almost identical to that reported in the literature.¹⁴

To observe the calorimetric change in ZIF-7 caused by CO₂ adsorption, DSC measurements were performed under CO₂ flow (Fig. 2a). Before the measurement, the temperature was increased from 25 to 200 °C at 20 K min⁻¹ under a nitrogen (N₂) flow rate of 50 mL min⁻¹ to desorb DMF remaining in the pores. The temperature was maintained at 200 °C for 10 min. The temperature was then cooled from 200 to -30 °C at 2 K min⁻¹ and subsequently heated from -30 to 200 °C under N₂ flow. No characteristic endothermic or exothermic peaks were observed under N₂ atmosphere (Fig. 2a, blue). After the same pretreatment as for the N₂ measurements, the gas was switched to CO₂ and the same DSC cooling/heating cycle was conducted. Under CO₂ atmosphere, an exothermic peak appeared around 45 °C during the cooling cycle (Fig. 2a, red). In the heating cycle, an endothermic peak was observed around 60 °C. These peaks were ascribed to calorimetric changes originating from the structural changes of ZIF-7-I associated with CO₂ adsorption and desorption. Next, TGA measurements were performed under similar conditions (Fig. 2b). No weight change was

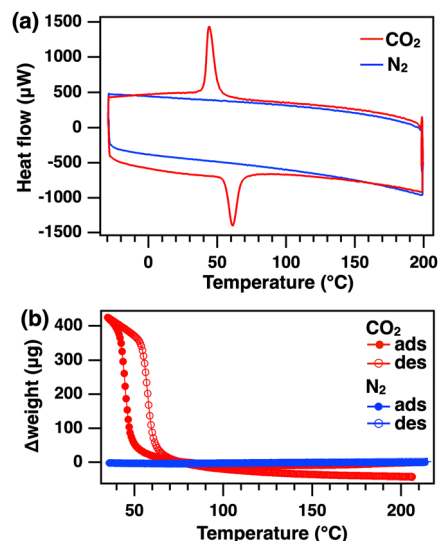


Fig. 2 (a) DSC curves of ZIF-7-I heated at 2 K min⁻¹ under CO₂ (red) and N₂ (blue). (b) Thermogravimetric analysis of ZIF-7-I under CO₂ (red) and N₂ (blue).

observed under N₂ atmosphere (Fig. 2b, blue). Under CO₂ atmosphere, weight changes attributed to CO₂ adsorption and desorption were observed during the cooling and heating processes, respectively. These weight changes occurred at temperatures similar to those of the exo- and endothermic peaks in DSC measurements (Fig. 2b, red). These results indicate that the observed calorimetric changes in DSC measurements were caused by the adsorption/desorption of CO₂ on ZIF-7-I.

The calorimetric changes of ZIF-7-I in CO₂ atmosphere were observed repeatedly in DSC measurements (five cycles, Fig. S2, ESI†). The peaks shifted to lower temperature and peak area increased with rising cycle number. Similar behavior was also observed in the TGA of ZIF-7-I in CO₂ (Fig. S3, ESI†). The first cycle showed a large difference between the weight before CO₂ adsorption and that after CO₂ desorption. As the number of cycles increased, this weight difference became smaller. In the first cycle, DMF and other gaseous substances remained in the pores of ZIF-7-I. The slight peak shifts observed in DSC curve and the increasing weight change in TGA curve during cycling are attributed to the gradual desorption of residual solvent and gas molecules. Judging from the negligible weight change upon completion of the fourth and fifth desorption cycles, the repeated weight change, which was accompanied by CO₂ adsorption, was almost complete by the fifth cycle.

The exothermic peak area of the adsorption process was calculated from DSC analysis to be -1.78×10^{-4} kJ. Based on the maximum adsorption amount in CO₂ adsorption/desorption measurements (Fig. 1b) and the isosteric heat of CO₂ adsorption of a typical MOF (25 kJ mol⁻¹), the theoretical adsorption heat was estimated to be -2.69×10^{-4} kJ. Judging from the difference between the observed exothermic heat in the DSC peak area and the predicted exothermic heat, the reaction heat was estimated to be $+0.91 \times 10^{-4}$ kJ. Since ZIF-7 undergoes a significant structural change from close form to open form upon CO₂ adsorption, the



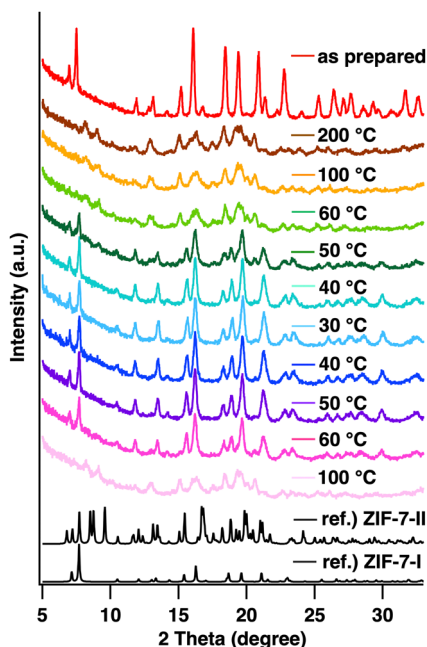


Fig. 3 *In situ* heating powder X-ray diffraction patterns of ZIF-7-I under CO₂ atmosphere. The reference XRD patterns of ZIF-7-I and II were simulated from cif files (CCDC 602541 and 2101613). Patterns were collected during cooling and heating in the following order: as-prepared (red), 200 °C (brown), 100 °C (orange), 60 °C (light green), 50 °C (green), 40 °C (turquoise), 30 °C (light blue), 40 °C (blue), 50 °C (purple), 60 °C (pink), and 100 °C (light pink).

origin of the estimated endothermic heat can be attributed to the reaction heat of the structural change. This indicates that the calorimetric change observed in DSC curve includes both the isosteric heat of CO₂ adsorption and the reaction heat due to the structural change of ZIF-7.

In situ heating powder XRD measurements were performed to confirm the structural changes observed in DSC measurements (Fig. 3 and Fig. S4, ESI†). When heated to 200 °C under N₂ flow, DMF in the pores of ZIF-7-I desorbed and its structure changed to ZIF-7-II. Then, the flow gas was switched to CO₂ and the sample was cooled to 30 °C. The diffraction pattern of ZIF-7-II changed to that of ZIF-7-I at 50 °C, which is consistent with the temperature of the exothermic peak observed in DSC measurements under CO₂. When the sample was heated to 100 °C in the second heating cycle, CO₂ was completely desorbed, and the diffraction pattern indicated that ZIF-7-I converted to ZIF-7-II again. The same measurements were performed under N₂ atmosphere (Fig. S4, ESI†). After the same pretreatment and cooling/heating cycle, no obvious change in the diffraction pattern of ZIF-7-II was observed, indicating that no structural changes occur under N₂ atmosphere. Therefore, the *in situ* heating powder XRD patterns revealed that the calorimetric and weight changes observed in DSC and TGA curves were caused by a gate-opening CO₂ adsorption/desorption process occurring in the crystalline ZIF-7 structure.

Because CO₂ adsorption on ZIF-7 is an equilibrium reaction, we attempted to determine the thermodynamic parameters of this process by changing the CO₂ concentration. For this purpose, DSC measurements were performed with different CO₂:N₂ ratios (Fig. 4a). As a pretreatment, a series of five

cooling/heating cycles was performed under CO₂. From the sixth cycle, the N₂:CO₂ ratio was changed for each cycle to decrease $p(\text{CO}_2)$. The exothermic peak caused by CO₂ adsorption shifted to the lower temperature side as $p(\text{CO}_2)$ decreased. Similarly, the endothermic peak caused by CO₂ desorption also shifted to a lower temperature as $p(\text{CO}_2)$ decreased. Since CO₂ adsorption on ZIF-7 is less likely to occur at low CO₂ partial pressures, the adsorption equilibrium temperature is expected to change to a lower temperature. This suggests that changes in $p(\text{CO}_2)$ directly influence the chemical equilibrium process of gate-opening CO₂ adsorption by ZIF-7.

Recently, Takiishi and Fujishiro²¹ reported the thermodynamic analysis of CO₂ adsorption by cubic perovskite-type BaFe_{1-x}In_xO_{3-δ}. They determined the standard reaction enthalpy and standard reaction entropy from TGA measurements at various $p(\text{CO}_2)$, and estimated the standard reaction Gibbs energy change ($\Delta_r G^\circ$) using the obtained parameters. Takamizawa *et al.*^{22,23} investigated the DSC behavior of one-dimensional coordination polymer crystals under CO₂ at low temperatures (0 to -35 °C). According to these reports, $\Delta_r G^\circ$ of CO₂ adsorption by ZIF-7-I was estimated from the exothermic peak temperature (T_p) in a series of DSC cooling curves at various $p(\text{CO}_2)$. Fig. 4b presents a plot of $\ln p(\text{CO}_2)$ against $1000/T_p$, which has a good linear correlation. The slope of the regression line of this plot gave $\Delta_r G^\circ = -15.1 \text{ kJ mol}^{-1}$. This experimentally determined value was consistent with the theoretical value obtained from DFT calculations in a previous study.¹⁷ Such thermodynamic parameters are clearly associated with the structural changes of ZIF-7 upon CO₂ adsorption/desorption.

The calorimetric changes in DSC at various CO₂ partial pressures (1.0–0.5 bar) are summarized in Table S1 (ESI†). In the case of lower CO₂ partial pressure, the observed calorimetric changes were slightly greater than that observed at 1.0 bar. As discussed above, since the calorimetric changes in the DSC curve are considered to include the exothermic heat of CO₂ adsorption and the endothermic heat of structural change, it is expected that there is a slight difference in the degree of structural change associated with partial CO₂ adsorption on the crystal surface or in the pore at lower CO₂ partial pressures.

We then measured the DSC behavior of another MOF to examine its ability to adsorb CO₂. Here, we focused on ZIF-9, which is a ZIF-7

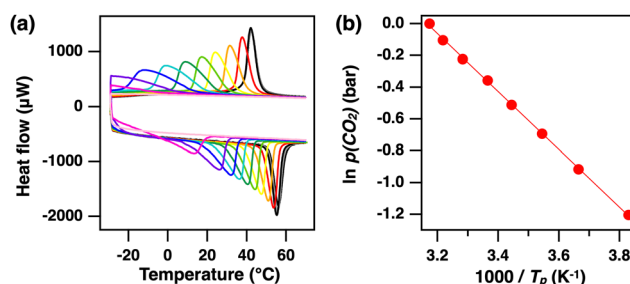


Fig. 4 (a) DSC curves of ZIF-7-I under ambient pressure and different CO₂/N₂ ratios of 1 (black), 0.9 (red), 0.8 (orange), 0.7 (yellow), 0.6 (yellow green), 0.5 (green), 0.4 (light blue), 0.3 (blue), 0.2 (purple), 0.1 (pink), and 0.02 bar (light pink). Above 70 °C is omitted for clarity. (b) Plot of the relationship between the CO₂ partial pressure and exothermic peak temperature (T_p) for ZIF-7-I.



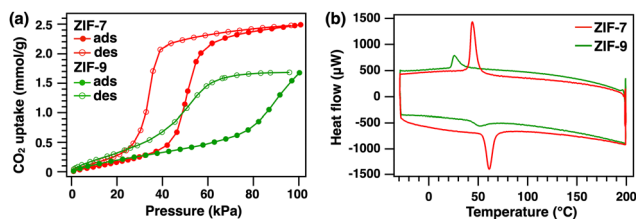


Fig. 5 (a) CO₂ adsorption isotherms of ZIF-7 (red) and ZIF-9 (green). (b) DSC curves of ZIF-7 (red) and ZIF-9 (green) heated at 2 K min⁻¹ under CO₂.

analog with a similar pore structure (pore diameter (phase II, closed state), ZIF-7: 0.29 nm, ZIF-9: 0.32 nm; pore diameter (phase I, open state), ZIF-7 and ZIF-9: 0.37–0.42 nm)^{13–17} that contains cobalt(II) ions rather than Zn²⁺ and also exhibits gate-opening CO₂ adsorption. ZIF-9 was synthesized in a similar manner to ZIF-7 (see the ESI† for details). Powder XRD measurements were performed, and the results were consistent with the formation of ZIF-9 (Fig. S5, ESI†). Fig. 5a shows a CO₂ adsorption/desorption isotherm of ZIF-9 along with one of ZIF-7 for comparison. The total CO₂ uptake for ZIF-9 was slightly lower than that for ZIF-7. The gate-opening CO₂ adsorption process of ZIF-9 was observed at a higher pressure than that of ZIF-7, suggesting that ZIF-9 has a lower CO₂ adsorption ability than that of ZIF-7. DSC measurements were performed for ZIF-9 under the same conditions as for ZIF-7 (Fig. 5b). Exothermic/endothermic peaks ascribed to CO₂ adsorption/desorption were observed for ZIF-9. However, the exothermic peak appeared at a lower temperature than that of ZIF-7, indicating that the CO₂ adsorption ability was lower for ZIF-9 than ZIF-7. Additionally, the DSC peak area for ZIF-9 (-0.71×10^{-4} kJ) was smaller than that for ZIF-7 (-1.78×10^{-4} kJ). This is consistent with the different CO₂ adsorption capacities of the two MOFs identified in their adsorption isotherms. Furthermore, since the estimated exothermic heat of CO₂ adsorption for ZIF-9 based on the adsorption isotherm is -2.00×10^{-4} kJ, the endothermic reaction heat due to the structural change is calculated to be $+1.29 \times 10^{-4}$ kJ, which value is higher than that of ZIF-7. This indicates the intrinsic energy change in the CO₂-induced structural transition of ZIF-9 is greater than that of ZIF-7. These results reveal that DSC under CO₂ is a useful thermodynamic analysis method for exploring the CO₂ adsorption process of MOFs and can be used to quantitatively evaluate various CO₂ adsorption materials.

In conclusion, DSC analysis of ZIF-7 under CO₂ atmosphere was conducted to evaluate its gate-opening CO₂ adsorption behavior. Calorimetric changes associated with the gate-opening structural changes of ZIF-7 occurred only under CO₂. *In situ* heating powder XRD and TGA of ZIF-7 under CO₂ atmosphere revealed that reversible weight and structural changes occurred during CO₂ adsorption/desorption. The thermodynamic parameters of the structural changes of ZIF-7 associated with CO₂ adsorption and desorption were calculated by analyzing the exothermic peak in DSC curves. The evaluation of gas adsorption capacity by DSC presented in this study for ZIF-7 and ZIF-9 is also applicable to other MOFs. The adsorption capacities of different types of MOFs can be compared based on peak areas and temperatures in DSC curves under CO₂ atmosphere. This means DSC is a useful thermodynamic analysis method for evaluating the gas adsorption/desorption

mechanisms of porous materials that undergo related structural changes. We are currently exploring the scope of DSC analysis with other gate-opening MOFs to better understand the thermodynamic parameters of the structural changes in MOFs that exhibit gate-opening gas adsorption behavior.

We acknowledge the financial support from the Japan Society for the Promotion of Science (JSPS) KAKENHI (Grant Numbers 19K05186 and 22K04857). We thank Natasha Lundin, PhD, from Edanz (<https://jp.edanz.com/ac>) for editing a draft of this manuscript.

Conflicts of interest

There are no conflicts to declare.

Notes and references

- H. Furukawa, K. E. Cordova, M. O'Keeffe and O. M. Yaghi, *Science*, 2013, **341**, 1230444.
- S. Kitagawa and R. Matsuda, *Coord. Chem. Rev.*, 2007, **251**, 2490–2509.
- A. Schneemann, V. Bon, I. Schwedler, I. Senkovska, S. Kaskel and R. A. Fischer, *Chem. Soc. Rev.*, 2014, **43**, 6062–6096.
- L. Chen, J. P. S. Mowat, D. Fairen-Jimenez, C. A. Morrison, S. P. Thompson, P. A. Wright and T. Duren, *J. Am. Chem. Soc.*, 2013, **135**, 15763–15773.
- J.-R. Li, R. J. Kuppler and H.-C. Zhou, *Chem. Soc. Rev.*, 2009, **38**, 1477–1504.
- N. Yanai, K. Kitayama, Y. Hijikata, H. Sato, R. Matsuda, Y. Kubota, M. Takata, M. Mizuno, T. Uemura and S. Kitagawa, *Nat. Mater.*, 2011, **10**, 787–793.
- S. Hiraide, Y. Sakanaka, Y. Iida, H. Arima, M. T. Miyahara and S. Watanabe, *Proc. Natl. Acad. Sci. U. S. A.*, 2023, **120**, 31.
- K. S. Park, Z. Ni, A. P. Cote, J. Y. Choi, R. Huang, F. J. Uribe-Romo, H. K. Chae, M. O'Keeffe and O. M. Yaghi, *Proc. Natl. Acad. Sci. U. S. A.*, 2006, **103**, 10186–10191.
- R. Banerjee, A. Phan, B. Wang, C. Knobler, H. Furukawa, M. O'Keeffe and O. M. Yaghi, *Science*, 2008, **319**, 939–943.
- M. R. Ryder, B. Civalieri, T. D. Bennett, S. Henke, S. Rudic, G. Cinque, F. Fernandez-Alonso and J.-C. Tan, *Phys. Rev. Lett.*, 2014, **113**, 215502.
- C. Gucuyener, J. V. D. Bergh, J. Gascon and F. Kapteijn, *J. Am. Chem. Soc.*, 2010, **132**, 17704–17706.
- P. Zhao, G. I. Lampronti, G. O. Lloyd, M. T. Wharmby, S. Facq, A. K. Cheetham and S. A. T. Redfern, *Chem. Mater.*, 2014, **26**, 1767–1769.
- X. Wu, M. N. Shahrak, B. Yuan and S. Deng, *Microporous Mesoporous Mater.*, 2014, **190**, 189–196.
- S. Aguado, G. Bergeret, M. P. Titus, V. Moizan, C. Nieto-Draghi, N. Bats and D. Farrusseng, *New J. Chem.*, 2011, **35**, 546–550.
- C. M. McGuirk, T. Runcevski, J. Oktawiec, A. Turkiewicz, M. K. Taylor and J. R. Long, *J. Am. Chem. Soc.*, 2018, **140**, 15924–15933.
- W. Cai, T. Lee, M. Lee, W. Cho, D.-Y. Han, N. Choi, A. C. K. Yip and J. Choi, *J. Am. Chem. Soc.*, 2014, **136**, 7961–7971.
- Y. Du, B. Wooller, M. Nines, P. Kortunov, C. S. Paur, J. Zengel, S. C. Weston and P. I. Ravikovitch, *J. Am. Chem. Soc.*, 2015, **137**, 13603–13611.
- P. Zhao, H. Fang, S. Mukhopadhyay, A. Li, S. Rudic, I. J. McPherson, C. C. Tang, D. Fairen-Jimenez, S. C. E. Tsang and S. A. T. Redfern, *Nat. Commun.*, 2019, **10**, 999.
- K. Kamali, S. Prasad, M. K. Sahoo, J. N. Behera, U. V. Waghmare and C. Narayana, *Inorg. Chem.*, 2022, **61**, 11571–11580.
- P. Zhao, G. I. Lampronti, G. O. Lloyd, E. Suard and S. A. T. Redfern, *J. Mater. Chem. A*, 2014, **2**, 620–623.
- K. Takiishi and F. Fujishiro, *J. Phys. Chem. C*, 2022, **126**, 20395–20400.
- S. Takamizawa, E. Nakata, H. Yokoyama, K. Mochizuki and W. Mori, *Angew. Chem., Int. Ed.*, 2003, **42**, 4331–4334.
- S. Takamizawa, E. Nakata, T. Akatsuka, R. Miyaki, Y. Kakizaki, H. Takeuchi, G. Maruta and S. Takeda, *J. Am. Chem. Soc.*, 2010, **132**, 3783–3792.

