# **Energy** Advances



View Article Online PAPER



Cite this: Energy Adv., 2023,

# LSNR:GO template for BSA interaction, photo and sonocatalytic reductions of fluorescent dyes in aqueous solutions†

Krishan Kumar, Bhargab Sahoo, Tara Chand Meghwal and Man Singh 60 \*

Lanthanide sulfide nanorods (LSNRs) were synthesized by fusing sodium (Na) metal with hexahydrate chloride salts of lanthanum (La), cerium (Ce), terbium (Tb), and holmium (Ho) (LnCl<sub>3</sub>·6H<sub>2</sub>O) with 4f<sup>0</sup>, 4f<sup>1</sup>,  $4f^9$ , and  $4f^{11}$  electrons ( $4f^{\#e}$ ), respectively. Na reduced LnCl<sub>3</sub>·6H<sub>2</sub>O to Ln(OH)<sub>3</sub> + 3H<sub>2</sub> as a crash reaction methodology (CRM). Aq-Ln(OH)<sub>3</sub> with H<sub>2</sub>S developed La<sub>2</sub>S<sub>3</sub>, Ce<sub>2</sub>S<sub>3</sub>, Tb<sub>2</sub>S<sub>3</sub>, Ho<sub>2</sub>S<sub>3</sub>, LSNRs, which were doped with graphene oxide (GO) at  $\sim$  60 for an LSNR:GO template (LGT), characterized with XRD, FT-IR, TGA/DTG, SEM, DLS, FESEM + EDS, C, H, N/S elemental analysis, Raman spectroscopy and survismetry. The 0.010 g% aq-LSNR and LGT were studied separately and with 0.010 g% (w/v) bovine serum albumin (BSA) from 283.15 to 315.15 K, and 0.002-0.010 g% @0.004 g% LSNR and LGT with 0.010% aq-BSA at 298.15 K. Their densities ( $\rho$ , g cm<sup>-3</sup>), viscosity ( $\eta$ , mPa s), sound velocity ( $\mu$ , m s<sup>-1</sup>), surface tension ( $\gamma$ , mN m<sup>-1</sup>), friccohesity ( $\sigma$ , s cm<sup>-1</sup>), isentropic compressibility ( $K_{s,\phi}$ , cm s<sup>2</sup> g<sup>-1</sup>), activation energy ( $E_a$ , J mol<sup>-1</sup>), and the thermodynamic parameters enthalpy ( $\Delta H$ , kJ mol<sup>-1</sup>), Gibbs free energy ( $\Delta G$ , kJ mol<sup>-1</sup>), and entropy ( $\Delta S$ , J mol<sup>-1</sup>) imply interactive activities with globular protein. LSNRs and LGT photocatalytically (PCR) and sonocatalytically (SCR) reduced methylene blue (MB), brilliant blue red (BBR), brilliant blue green (BBG) and methyl orange (MO). LGT recycled wastewater containing fluorescent dyes in  $\sim$ 60 min while GO did it in 120 min and LSNRs in 24 h under sunlight (SL). Sonicated dyes with aq-LGT at 28 kHz for 5, 10, 20, 30, 60 min were reduced with 40-90%  $\Phi$  compared to 98% with PCR in 1st order kinetics. The physicochemical properties (PCPs) of LSNR and LGT predicted their structural activities with reusability.

Received 7th October 2022, Accepted 9th December 2022

DOI: 10.1039/d2ya00274d

rsc.li/energy-advances

# 1. Introduction

Lanthanide and GO individually have been widely used for research and industrial applications, but their interfaces have not yet been reported.<sup>1,2</sup> Therefore, developing their LGT has been a novel research project and it was found to be relentlessly stable, reusable, and efficient photocatalyst to split water expeditiously compared to GO and LSNRs alone.<sup>3,4</sup> Synthesizing new photocatalyzing nanomaterials for reducing wastewater effluents in sunlight is a biological and natural requirement for recycling. PCR splits wastewater into H<sub>2</sub> and O<sub>2</sub> gases, and also recycles its fluorescent effluents and transitional metals to collect usable water.<sup>5</sup> For decades, the focus has been on developing photocatalyzing metallic organic frameworks, metallic biocomplexes, and bimetallic complexes using transitional metals,

School of Chemical Sciences, Central University of Gujarat, Sector-30(382030), Gandhinagar, India. E-mail: Mansingh50@hotmail.com, Krishan8053649040@gmail.com, rajatvlrc@gmail.com, tarachandmeghwal9660@gmail.com

as reported in previous studies.<sup>5,6</sup> Transition metals are comparatively easy to handle, and their crystalline water molecules are removed at NTP unlike the 800-1300 for lanthanide metals that hinders chemical processes.<sup>7</sup> The crystalline 6H<sub>2</sub>O of LnCl<sub>3</sub>·6H<sub>2</sub>O is not detached by dissolving it in water at NTP, though, its Cl<sup>-</sup> dissociate and respond to AgNO<sub>3</sub> but are unable to form Ln<sub>2</sub>S<sub>3</sub>. As a result, researchers have been heating LnCl<sub>3</sub>· 6H<sub>2</sub>O up to 800-1300 to prepare Ln<sub>2</sub>S<sub>3</sub> without understanding the contraction effect of 4f# electrons8,9 (Table 1). The 6s electrons shield a nuclear charge that affects the activities of 4f #e and the water dipoles are strongly bound with the nuclear charge, restricting lanthanides interacting with H2S for LSNR preparation and applications in the areas of photocatalysts, and adsorbents. The problem of preparing Ln<sub>2</sub>S<sub>3</sub> by heating to detach a crystalline 6H2O from LnCl3·6H2O due to the energy barrier for electronic reduction of 6H2O is resolved by allowing a reaction with sodium metal at 50 with LnCl<sub>3</sub>·6H<sub>2</sub>O. The 3s<sup>1</sup> electron of sodium metal reduces 6H<sub>2</sub>O to H<sub>2</sub> and Ln(OH)<sub>3</sub> (Fig. 3). The idea of synthesizing LSNRs and LGT for use as photocatalysts and adsorbents has never been initiated, despite huge industrial potential in areas of effluent disposal of

<sup>†</sup> Electronic supplementary information (ESI) available. See DOI: https://doi.org/ 10.1039/d2ya00274d

 $\begin{tabular}{lll} \textbf{Table 1} & \textbf{Comparative} & \textbf{study} & \textbf{of} & \textbf{lanthanide} & \textbf{sulfide} & \textbf{preparation} & \textbf{with} \\ \textbf{temperature} & & \begin{tabular}{lll} \textbf{Table 1} & \textbf{Sulfide} & \textbf{preparation} & \textbf{with} \\ \textbf{Table 2} & \textbf{Table 3} & \textbf{Table 3} & \textbf{Table 3} \\ \textbf{Table 3} & \textbf{Table 4} & \textbf{Table 3} & \textbf{Table 3} \\ \textbf{Table 4} & \textbf{Table 4} & \textbf{Table 5} & \textbf{Table 5} \\ \textbf{Table 5} & \textbf{Table 6} & \textbf{Table 6} & \textbf{Table 6} \\ \textbf{Table 6} & \textbf{Table 6} & \textbf{Table 6} \\ \textbf{Table 6} & \textbf{Table 6} & \textbf{Table 7} \\ \textbf{Table 7} & \textbf{Table 7} & \textbf{Table 7} \\ \textbf{Table 7} & \textbf{Table 7} & \textbf{Table 7} \\ \textbf{Table 7} & \textbf{Table 7} & \textbf{Table 7} \\ \textbf{Table 7} & \textbf{Table 7} & \textbf{Table 7} \\ \textbf{Table 7} & \textbf{Table 7} & \textbf{Table 7} \\ \textbf{Table 8} & \textbf{Table 7} & \textbf{Table 7} \\ \textbf{Table 8} & \textbf{Table 7} & \textbf{Table 7} \\ \textbf{Table 8} & \textbf{Table 7} & \textbf{Table 7} \\ \textbf{Table 8} & \textbf{Table 8} & \textbf{Table 7} \\ \textbf{Table 8} & \textbf{Table 8} & \textbf{Table 8} \\ \textbf{Table 9} & \textbf{Table 9} \\ \textbf{Tab$ 

This study         50–60         Crash reaction methodology           1         800–1000         Hydrothermal process         26 and 32           2         650–750         Solid phase reaction         8 and 33           3         500–650         Solvothermal         34 and 35           4         100–500         Hydrothermal reaction         21 and 36           5         500–1000         Solvothermal         37 and 38	Sr. no.	Temp. (°C)	Method	Ref.
	1 2 3 4	800-1000 650-750 500-650 100-500	Hydrothermal process Solid phase reaction Solvothermal Hydrothermal reaction	8 and 33 34 and 35 21 and 36

infectious and hazardous biomedicals, and also, for solving the most tedious and hazardous chemical processes. LSNRs and LGT act as renewable energy storage systems and dopants with pharmaceuticals, digital inks, textiles, integrated circuits (ICs), and heat sensors. 5,6,10,11 PCPs, PCR, and SCR predict their structural abilities to photocatalyzed and solubilize the biomolecules as BSA LSNR and LGT nanomaterials could also be doped and blended with a fine powder of waste plastics to functionalize and recycle them. 12 LSNRs and LGT could structurally align GO and polymer composites for wider applications as supercapacitors and metallic nanocatalysts. 8,9,13 The quantized interface (4f  $^{\text{#e}} \sim \text{FE}$ ) could split light and water due to mutually resonating energy transfer for separating light.<sup>14</sup> Thereby, LGT depicted as  $4f^{\#e} \sim FE$  could replace the expensive prisms used for splitting light for the faster photocatalysis of industrial effluents. 15 LSNRs could also be doped with proteins, chitosan, carbohydrates/cellulose as biotemplates with a 4f<sup>#e</sup> ~ FE interface for biophysical applications. <sup>16</sup> LGT with 4f #e ~ FE as an energy simulator could trap solar radiation for bio and piezoelectric sensors and minimize light scattering, and quantum confinements within optoelectronics.3,17 Unlike MOFs of transition metals, the lanthanide organic framework (LOF) could fill a need for applications with higher frequency of recycling and energy storage. 18 The functional edges (FE) and  $\pi$  conjugations of GO with photons  $(h\nu)$  generates a wavefunction  $(\psi)$  in different phases  $(\varphi)$ , but the 4f<sup>#e</sup> ~ FE interface photocatalyzed a dye in a shorter time with an intensified redox cycle (ROC).15-17 LSNRs and LGT could affect salt bridges, electrostatics, and the peptide bonds of a protein by generating thermodynamic parameters tuned by 4f #e electrons. 18,19 GO as an  $h\nu$  receptor has been a key factor in the fluorescence and photoluminescence of LGT. 20,21 The 4f #e align GO with the lowest permittivity ( $\varepsilon$ ) and quantum energy barrier (QEB) without causing electron-electron collisions. LGT with a highly resonating frequency of 4f#e and GO together attain outstanding optical, luminescence, and magnetic properties with extensive screening for diagnostic applications in the biological domain.<sup>22</sup> LGT with balanced hydrophobicity and hydrophilicity could boost drug delivery, fluorescent probes, bioimaging, biosensors, and contrast agents in MRI scanning interaction with globular protein, but no study has been reported yet.<sup>22-26</sup> The 4f#e ~ FE interface as a new framework is aimed at separating out hazardous materials from biomedical waste effluents. PCPs of LGT suggest intensification of molecular interactions to recycle wastewater via hydrophobic and hydrophilic activities. The morphology, crystallinity, and redox activities of LSNRs have not yet been reported.<sup>26-28</sup> The hydrophobic and hydrophilic interactions of Ln(NO<sub>3</sub>)<sub>2</sub>:6H<sub>2</sub>O with BSA were extended to LSNRs and LGT in previous studies.<sup>29</sup> LSNRs and LGT could not unfold aq-BSA except by solubilising by using partial surface charges to interact with LGT. Aq-LSNR and LGT solutions align the dipoles of water that affects the adherence and cohesion of a solution with higher PCR and SCR activities interfaced with PCPs for understanding the OEB. The multifunctional LGT gets fitted with electrostatic dipoles, gaining a photon trapping ability for PCR and SCR activities. BSA interactions with partially soluble LSNRs and LGT could tune the activities of hormones and enzymes due to S2- with temperature.30 The shear stress implies a suspended BSA aligned in solution, so survismetry explores their alignments.<sup>31</sup> The aligned dipoles in aqueous medium catalyse the reduction of dyes via photonic and sonication energies. Therefore, the rheology, surface energy, and thermodynamic parameters have been determined as a prerequisite database for PCR and SCR activities.

# 2. Experimental section

#### 2.1. Materials

Gt,  $\rm H_2SO_4$  (99.9%),  $\rm H_3PO_4$  (85%), KmnO<sub>4</sub>,  $\rm H_2O_2$ , HCl, chloride hexahydrate salts of lanthanum, cerium, terbium, holmium, sodium metal, BSA, iron sulfide, methylene blue, brilliant blue R, brilliant blue G, and methylene orange were procured from Sigma Aldrich; petroleum ether (40–60) from Sisco Research Laboratory (SRL) India, and absolute alcohol from SCVUS Mandli India, and the solutions were prepared in Milli-Q water weighed using a  $\pm 1 \times 10^{-6}$  balance.

### 2.2. Preparation of GO and GtO from Gt

Graphite oxide (GtO) and GO were synthesized from Gt flakes with KmnO<sub>4</sub> and H<sub>2</sub>SO<sub>4</sub>: H<sub>3</sub>PO<sub>4</sub> in 8.5: 1.5 ratio poured dropwise into an RB flask placed on an ice bath containing Gt and KmnO4 in 1:6 ratio. These were refluxed at 55 @650 rpm in an oil bath for 12 h to reach RT. To obtain pure GtO, ice-cold water was mixed with a reaction mixture on adding 3 mL H<sub>2</sub>O<sub>2</sub> (30%) and stirred for 10 min to cease a reaction.5 The resultant product was centrifuged at 8000 rpm for 10 min and washed with Milli-Q water, HCl and absolute alcohol until the supernatant attained pH 7. AgNO3 and BaCl2 tests ensured complete removal of NaCl and Na2SO4 after 5 washings, and a brownish sediment was coagulated with petroleum ether. It was extracted to a Petri dish for drying at 60 for 24 h, a brown-colour solid GtO was obtained with good reproducibility and feasibility after repeating the procedure several times. GtO was exfoliated by sonicating at 28 kHz for 3 h to disrupt the ABAB sheet stacking of the Gt layers. GtO disperses brownish GO homogeneously39 (Fig. 1 and 2). The reaction possibly occurs as follows:



Fig. 1 GO preparation from Gt flakes.

Fig. 2 LSNR preparation

#### GO functionalization

$$6KMnO_4 + 3H_2SO_4 + 4H_3PO_4$$

$$\rightarrow 3K_2SO_4 + 2Mn_3(PO_4)_2 + 9H_2O + 15[O]$$

and

$$KMnO_4 \rightarrow K^+ + Mn^{7+} + 4O^{2-},$$

$$4O^{2-} \rightarrow 2O_2 + 8e^-,$$

$$Mn^{7+} + 5e^- \rightarrow Mn^{2+},$$

$$Gt + 15[O] \rightarrow GtO,$$

$$GtO + \cong \rightarrow GO$$

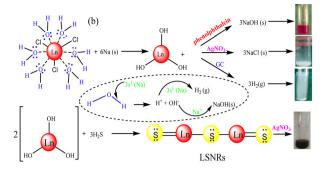
### 2.3. Crash reaction methodology (CRM) for LSNR synthesis

Crushed Na metal taken in a boiling tube was heated to 50 °C for  $\sim 1$  min, then  $LnCl_3 \cdot 6H_2O(s)$  was added in 1:6 ratio. It sparked the production of  $H_2$  gas and  $Ln(OH)_3$  within nanoseconds, and the mechanism is denoted CRM. To solubilize  $Ln(OH)_3$ , water in 1:4 ratio was added to homogenize the solution (Fig. 1). An excess of  $H_2S$  gas as a source of  $S^{2-}$  was passed (10 min) to react with  $Ln^{3+}$  to produce  $Ln_2S_3$ . The  $H_2$  gas, NaCl and NaOH were confirmed with gas chromatography, AgNO<sub>3</sub> and phenolphthalein, respectively (Fig. 3). The  $3s^1$  Na reacted with  $6H_2O$  to disrupt its dipolar linkage with  $Ln^{3+}$ , generating  $H^+$  that received  $3s^1$  as  $2H^+ + 2e \rightarrow H_2$  (Fig. 3) as follows:

$$LnCl_3 \cdot 6H_2O + 6Na \rightarrow Ln(OH)_3 + 3NaCl + 3NaOH + 3H_2 \uparrow$$
  
 $2Ln(OH)_3 + 3H_2S \rightarrow Ln_2S_3 + 6H_2O (LSNR)$ 

A temperature of 50  $^{\circ}$ C ignited  $3s^{1}$  to gain adequate oscillations which simultaneously interfaced with  $H_{2}O$  to dissociate it. The product was washed with Milli-Q water 4/5 times to remove NaCl and NaOH. The whitish to greenish colour confirmed LSNRs which with AgNO $_{3}$  had formed black precipitate implying  $S^{2-}$  (Fig. 3).  $Ln_{2}S_{3}$  was washed 2–3 times and dried at 50  $^{\circ}$ C for 2 h in a vacuum oven.

La<sub>2</sub>S<sub>3</sub> (S<sup>2-</sup> = Ln<sup>3+</sup>-Ln<sup>3+</sup> = S<sup>2-</sup>) with 4f<sup>0e</sup> have developed cohesive forces and with less kinetic energy have aligned to form linear rods (Fig. 4), although the white patches imply thick nanorods (NRs) due to linearly aligned S<sup>2-</sup> and Ln<sup>3+</sup>. The S<sup>2-</sup> oscillate to align LSNRs and 4f <sup>#e</sup> in different  $\varphi$ . The double bonds of S<sup>2-</sup> = Ln<sup>3+</sup>- and -Ln<sup>3+</sup> = S<sup>2-</sup> and both double bonds of S<sup>2-</sup> = Ln<sup>3+</sup>-S<sup>2-</sup> = S<sup>2-</sup> align linearly with the spatial lattice as NRs. The oscillation of the double bonds of S<sup>2-</sup> = Ln<sup>3+</sup>-S<sup>2-</sup>-S<sup>2-</sup>



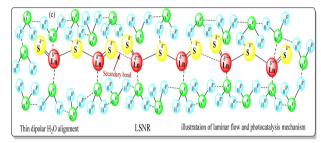


Fig. 3 (a) CRM with evolution of  $\rm H_2$  gas, (b) LSNR confirmatory test and (c) alignment.

affects the electronic activities of  ${\rm Ln^{3+}}$  and  ${\rm S^{2-}}$  due to electrostatic interactions at the terminals. There are no possibilities of charge oscillations as their  $\psi^2$  is attained and  ${\rm Ce_2S_3}$  (4f  $^{1\rm e}$ ) has created a cage-like structure compared to  ${\rm La_2S_3}$  with maximum oscillations in different  $\varphi$ . Additionally,  ${\rm Tb_2S_3}$  with 4f  $^{9\rm e}$  shows a finer distribution for developing a sheet-like arrangement.  ${\rm Ho_2S_3}$  with 4f  $^{11\rm e}$  shows perfect NRs on aligning the  ${\rm S^{2-}}$  with the conduction band (CB) (Fig. 4). The LSNR has one dimension (1D) due to having a wavefunction of the unoccupied state of the lone pair of electrons of the sulfur atom including its

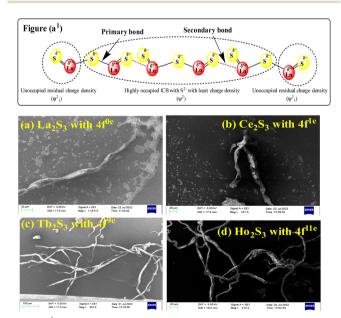


Fig. 4 (a¹) Preferred interaction for LSNR formation and SEM images (a)  $La_2S_3$  (b)  $Ce_2S_3$  (c)  $Tb_2S_3$  (d)  $Ho_2S_3$ .

negatively charged domain facilitated by 4f #e (Fig. 4a1). The secondary bond of Ln<sup>3+</sup> and S<sup>2-</sup> static ions help to equipartition their residual electronic charges through quantizing their wavefunctions, generating communication for their spatial alignment as a state. Thus, their eigenenergy and eigenfunctions generate a  $\psi(x, t) \times \psi^*(x, t)$  density of energy state (DOS), which was elucidated with elemental diffraction X-ray spectroscopy and CHNS/O analysis. Our previous study of GO, Pr<sub>2</sub>S<sub>3</sub>-GO, Sm<sub>2</sub>S<sub>3</sub>-GO, Eu<sub>2</sub>S<sub>3</sub>-GO, and Gd<sub>2</sub>S<sub>3</sub>-GO used HR-TEM images with 20 nm range to infer 1D nanorods. 40 Additionally, the HR-TEM analysis of Gd<sub>2</sub>O<sub>3</sub> doped with Eu<sup>3+</sup> and capped with PVP functionalised with HSA maintained the nanorod dimension reported in our earlier study. 41 A similar methodology to that of LSNR and LGT was extended to the present study where the 4f #e of LSNR and FE of GO have been coordinated through their respective  $\psi_{\text{#4fe}}$  and  $\psi_{\text{FE}}$  to initiate the  $\psi_{\text{#4fe}}(x_1, t) \times \psi_{\text{FE}}(x_2, t)$  DOS. The  $x_1$  and  $x_2$  positions of the 4f<sup>#e</sup> and FE, respectively, differ infinitesimally to generate  $(q^{\#4fe})$ ,  $q^{\text{FE}}/4\pi\varepsilon r^2$ , as the intermixing distance r is approachable. These highly inclusive electronic states, as per the Born-Oppenheimer approximation, optimize and simulate the electronic and nuclear charges, leading to coordination via the S2- ions of an LSNR with consecutive LSNRs. Thereby, the changes occur at nanometre distance, causing stacking LSNRs to stack linearly, as in Fig. 4a-d. SEM images of La<sub>2</sub>S<sub>3</sub>, Ce<sub>2</sub>S<sub>3</sub>, Tb<sub>2</sub>S<sub>3</sub>, and Ho<sub>2</sub>S<sub>3</sub> imply linear rods. However, they have formed a bundled state on combining close to each other, but the dimension remains broadly 1D.

#### 2.4. LGT synthesis via sonothermal process

GO (mg mL<sup>-1</sup>) was sonicated for 3 h to disperse the sheets in water to generate FE (Fig. 5a). LSNR powder in GO solution with 1:5 ratio (dopant: matrix) was sonicated for 30 min at 60 and cooled to RT (Fig. 5). GO sheets with a floppy structure engaged with LSNRs for doping. (Fig. 5d). After the LGT was washed with H<sub>2</sub>O to remove LSNRs weakly bonded with GO, it was put through a centrifuge for 8 min at 8000 rpm. The LFT was dried in a vacuum oven for 2 h at 50 as shown below:

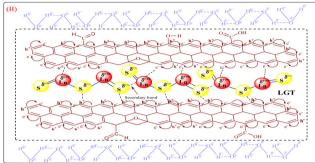
$$GO + \cong \rightarrow GO^*, GO^* + Ln_2S_3 \rightarrow Ln_2S_3:GO (LGT)$$

Spatial alignments and the geometrical appearance of LGT are shown in Fig. 5.

#### 2.5. Characterizations

UV/Vis spectra were recorded with a Shimadzu UV1800 double beam spectrophotometer and X-ray diffraction (XR) with a Bruker Advance-D8 diffractometer with Cu Kα radiation ( $\lambda$  = 0.15406 nm, 40 kV, 40 mA). TGA was conducted in an  $N_2$  environment on an EXSTAR TG/DTA 7300 to analyze thermal stability. C, H, N, and S analysis was undertaken with a Euro-vector EURO EA 3000, and FESEM + EDS with a JEOL ISM 7100F. The Raman spectra were recorded with a Horiba HR 800 instrument with a 480 nm Ar laser. Functional groups (FGs) were analysed with an FT-IR spectrometer (PerkinElmer, Spectrum 65), H<sub>2</sub> evolution with a gas chromatograph (Perkin-Elmer, Clarus 680), and PCPs and density with a Mansingh survismeter (Cal.06070582/1.01/C - 0395,NPL,GOI) and Anton par Densimeter-5000, respectively.





(I) LGT preparation and (II) alignment

# Results and discussion

LnCl<sub>3</sub>·6H<sub>2</sub>O with a stronger nuclear charge can hold 6H<sub>2</sub>O and detach 3Cl<sup>-</sup> in aqueous solution in response to AgNO<sub>3</sub>, Na<sup>+</sup>Cl<sup>-</sup> with H2S gas developed Na2S but was unable to develop Ln2S3. Heating LnCl<sub>3</sub>·6H<sub>2</sub>O could not thermally induce electronic spin to overcome a nuclear charge being applied on 6H2O. LnCl3. 6H2O reacted directly with H2S gas but did not form LSNRs. The NaOH added in aq-LnCl<sub>3</sub>·6H<sub>2</sub>O gives NaCl and free S<sup>2-</sup> to capture Ln<sup>3+</sup> by weakening their inner hydration sphere (IHS). The weaker IHS of LnCl<sub>3</sub>·6H<sub>2</sub>O attracts free S<sup>2-</sup> and forms a cloudy cluster around Ln<sup>3+</sup> (Fig. 6).

Previous researchers could not analyse the electronic bonding of 6H<sub>2</sub>O with LnCl<sub>2</sub> in areas of lanthanide chemistry (Table 1). LGT with 4f <sup>#e</sup> with a resonating energy tuned the biophysical and biochemical activities of BSA for an interfacial study extendable to haemoglobin via the holding/releasing ability of O<sub>2</sub> and CO<sub>2</sub> gases. Solute and solvent interaction affect various activities, where PCPs of LSNRs and LGT imply alignment to enhance the PCR and SCR. Survismetry of GO, LSNRs, and LGT in aqueous solutions through a uniform capillary has elucidated their incompressible dipoles interfaced with PCR and SCR activities. The aligned dipoles of LGT could have minimized the localized collisions of e<sup>-</sup> and h<sup>+</sup> holes by avoiding their recombination due to a robust friccohesity resonating energy transfer (FRET). In the reaction  $LnCl_3 \cdot 6H_2O \rightarrow Ln^{3+} + 3Cl^-$ , in addition to the  $6H_2O$ 



Fig. 6 Direct sulfurization with aq-NaCl and LnCl<sub>3</sub>·6H<sub>2</sub>O.

molecules, infinite dipoles of H<sub>2</sub>O align with Ln<sup>3+</sup>, developing the strongest electrostatic interactions. These activities spontaneously favour the generation of electron-deficient  $\psi_{\mathrm{Ln}^{3+}}$  and  $\psi_{\rm OH_2}$  clouds. These clouds are highly stable even when dissolved in H<sub>2</sub>O, though the 3Cl<sup>-</sup> anions get dissociated, which further exceptionally reorients the H<sub>2</sub>O dipole to optimize the negatively charged clouds of H2O with maximum hybridization. The Clhad hurdled the alignments of H<sub>2</sub>O to LnCl<sub>3</sub>·6H<sub>2</sub>O, but after the removal of the 3Cl<sup>-</sup>, an infinite number of H<sub>2</sub>O had surrounded the Ln<sup>3+</sup>, which interfered with the bonding the S<sup>2-</sup> supplied on passing H<sub>2</sub>S. These activities are supported with 4f<sup>#e</sup> of lanthanide due to 5d and 6s suborbitals sharing a nuclear charge that weakens the attraction of the 4f#e for a nuclear charge. The dissociated Cl<sup>-</sup> gave a positive test with AgNO<sub>3</sub> as a white AgCl cloud while it was noted that strongly hydrated Ln<sup>3+</sup> could not respond to H<sub>2</sub>S or Na<sub>2</sub> (Fig. 6). The chemical species Na<sup>+</sup> + S<sup>2-</sup> and H<sup>+</sup> + S<sup>2-</sup> obtained on dissociation of the Na<sub>2</sub>S and H<sub>2</sub>S respectively could not attract the hydrating H<sub>2</sub>O dipoles. As the dipoles were electronically bonded with Ln3+H2O-Ln3+-6H2O and confined to quantum dots (QD-Ln-NHS) highly intensified with a highest energy. This energy acted as non-permissible reaction so the S<sup>2-</sup> could not penetrate the QD-Ln-NHS. Hence, the large S2- could not penetrate QD-Ln-NHS, so aq-Na2S could not develop Ln<sub>2</sub>S<sub>3</sub>. The S<sup>2-</sup> could not cause adequate spin, rotational, vibrational, or translational movement compared to a 3s1 electron due to the least size that could enter the QD-Ln-NHS to reduce the  $H_2O$  (Fig. 3a). The  $2H^+(2H^+ + 2e^- \rightarrow H_2) + 2OH^$ and Na<sup>+</sup> expeditiously develop NaOH. Another negative factor in using Na<sub>2</sub>S is that it is dangerous for the H<sub>2</sub>O molecule to develop a cationic hydration sphere. The Na metal with a 3s1 electron is not a danger, but rather is a highly favourable mechanism where the spin of the  $3s^1$  electron gains a robust  $\psi$  to overcome a quantum tunnelling effect to separate out the hydrating H2O dipoles. Hence, there are no electronic or vibrational energies to facilitate the Na-3s1 electron reaching the QD-Ln-NHS unlike S<sup>2-</sup>. The energy which is expeditiously released during the reaction excites the electron of the hydrating H2O molecule of LnCl<sub>3</sub>·6H<sub>2</sub>O and separates it out. A precaution which is maintained to synthesize LSNRs using Na-3s<sup>1</sup> is that small amounts of both the reagents were taken. Additionally, Na2S acquires energy from the reacting system to dissociate the 2Na<sup>+</sup> + S<sup>2-</sup>, which attracts the cation (CHS) and anion (AHS) hydration spheres, weakening the activities of the H<sub>2</sub>O medium to affect the alignment of the CHS and AHS around QD-Ln-NHS, to not only immobilize the structural activities of the H<sub>2</sub>O medium but also remove the chemical potential ( $\mu$ ) to develop Ln<sub>2</sub>S<sub>3</sub> as  $a = e^{\mu}/RT$ activities. Several reactions with Na2S, H2S, K2S, Ca2S, and others were conducted which could not produce Ln<sub>2</sub>S<sub>3</sub>. The Na-3s<sup>1</sup> electron is easily accessible to the  $H^{\delta}$ + population and could not reduce  $Ln^{3+}$  to  $Ln^{0}$  metal, avoiding the  $Ln^{3+} + 3s^{1} \rightarrow Ln^{0}$ reaction and instead consuming the  $H^{\delta+}$  charges. The present mechanism is safer as almost all the H2O molecules are either strongly hydrogen bonded where a major chunk of H2O is hydrating Ln<sup>3+</sup> and 3Cl<sup>-</sup>. Hence, the Na metal was warmed up to 50-55 to react with Ln3+. Had there been free H2O, there would have been a stronger reaction. Detailed studies of the CRM were

reported in our previous study.  $^{40}$  Lots of trials were made to synthesize the LSNRs by avoiding a direct heating to 800–1300, as reported in earlier studies (Fig. 6 and Table 1). Developing the infrastructure for generating 800–1300 is very difficult and dangerous, unlike using Na metal. The flame shown in Fig. 3a belongs to the  $\rm H_2$  gas which was produced during the reaction. Of course, by developing pressure within a few seconds it is diffused. The  $\rm H_2$  gas was intensively confined to a flame as we wanted to confirm it with a popping sound and a yellowish flame. However, provision was made to generate  $\rm Ln^{3+}$  by allowing contact with a small amount of Na metal by passing it through a conveyor belt to the reaction chamber and attracting the resultant  $\rm Ln^{3+}$  to  $\rm H_2S$  gas to develop  $\rm Ln_2S_3 + \rm H_2$ . Hence, the conveyor belt system is highly scalable.

LGT and LSNRs with a smaller surface area induce stronger interactions between the  $\delta^-$  and  $\delta^+$  aligned dipoles of H<sub>2</sub>O. The 4f<sup>#e</sup> of LGT oscillates with  $\psi$  and minimized  $\varepsilon$  for expeditious PCR and SCR activities of dyes, causing the uniform alignment shown with PCPs. The electrostatic dipoles of H<sub>2</sub>O interact with monodispersed aq-LSNR and aq-LGT, so a laminar flow was studied from 0.002 to 0.10 g% @ 0.004 g% to align them (Fig. 13). The and values do not vary on increasing the g% of LSN and LG, as they could act in ordered alignment as a supercapacitor (Tables 5 and 6). The water dipoles and the aligned LSNR:GO develop a pendant drop number (pdn) saturated at 298.15 K with new surfaces (Fig. 14b). LGT formed a large size drop with increasing g% due to alignment with a more strongly adhered solution and multiple nanohydration spheres (NHSs) that isolated their cohesive forces (Fig. 6). Aq-LGT adheres to a solid glass surface by developing a pdn in continuous flow, cohesively holding a solution with an adequate cohesive force of adhered water to LGT and surrounding water molecules. This action develops the size and length of a pdn due to stronger adhesive and cohesive forces of H2O to LGT and with itself. The lower surface area of a drop implies H2O adhered to LGT forming an NHS around the LGT, weakening the H2O-H2O electrostatic interactions (Fig. 7). GO, LSNR, and LGT act as competent photocatalysts to PCR and SCR fluorescent dyes in aqueous solution. On receiving  $h\nu$ , the surface area and surface energy, with their realignment and reorientation in the photocatalyzing liquid mixture, have become the most critical interfaces. The photocatalysts on receiving the  $h\nu$  haphazardly collide, increasing the probability of holes recombining, so the  $h\nu$  may not generate effective quantum dots to cause  $h\nu$  to excite the electron. Hence surface enhanced Raman spectroscopy (SERS) highlights the surface activities reported in our previous study.3,40 Additionally, due to the surface energies, the molecules develop NHSs, which could combine via the cohesive force that develops pdns of different sizes, affecting the photocatalysis. Thereby, the compositions of GO, LSNRs, and LGT generate the surface activities which influence the width of the pdn with a lower surface area. Fig. 7 shows that the 0.002, 0.004, 0.006, and 0.010 g% LSNRs respectively increase the size of the pdn. The 0.010% sample has a comparatively longer pdn due to stronger cohesive forces with slightly milder adhesive forces with a lower tip of the hanging capillary with the controlled

Increasing size of drop with @g% due to lower surface area

pressure unit (Fig. 7). This mechanism of milder adhesion, with either a concentrated photocatalyst or a dye with comparatively stronger cohesive forces, by itself increases the QEB to hamper the photocatalysis. The pdn decreases as the cohesive forces are strengthened with increasing composition, which could reduce the quantum yield (Fig. 7).

GO has been synthesized and widely characterized with XPS, XRD, TGA, HR-TEM, SEM, and EDX reported in our previous studies. 42 Widely and vividly reported studies suggested that GO has a 2D sheet structure with hexagonal units with moderate VWF. GO has been functionalised by generating oxygenated FGs, which further realigned the 2D GO sheet with slightly different intersheet VWF causing the lowest relaxation time. These structures have been extensively studied with stretching frequency and bending frequency (Fig. 9 and Fig. S2.9, ESI†). The FT-IR analysis for LSNRs and LGT individually elaborated the spatial geometry with specific rotational, vibrational, and electronic motions integrated as stretching and bending frequencies. The C-C-C atomic domains of LGT for 1900-3000 cm<sup>-1</sup> are almost linear compared to a GO matrix, which indicates the activities of the FGs. These activities were minimized and FGs were used in doping the LSNRs. But dopants with specific doping activities to tune the C-C-C atomic domains are lacking for individual LSNRs (Fig. S1.1, ESI†). The S-S atomic oscillations are depicted with the pattern of split of stretching and bending frequencies where the integrated 2D GO sheet with highly stable C-C-C atom domain that could not split the stretching and bending frequencies. These fundamental patterns exclusively elucidate the structure of GO, LSNRs, and LGT. Intrinsic friccohesity correlates the stretching and bending frequencies of GO, LSNRs, and LGT as the intrinsic electronic configurations of each adjoining atom rotate, vibrate, and translate with specific oscillations. These generate specific patterns of electron-electron repulsion, electronic collision, and mean free path, as per Lorentz activities. These oscillations generate a specific pattern in TGA, DTG, and DSC (Fig. 10). GO generates a single sharp line reflecting its single lattice structural pattern. However, the single sharp line is transformed into different patterns with the LSNR spatial lattice, implying that the 4f #e of LSNRs seem to coordinate for mutual optimization. Hence the specific patterns of the LGT differ from those of GO with faster photocatalyzing activities. These changes imply the participation of the FGs of GO and the 4f #e of the LSNRs due to their structural alignments. FT-IR and TGA analyses have extensively elaborated the originality and functionality of their structures, developing specific patterns of wavevector and lattice, as depicted by the XRD pattern (Fig. 8).

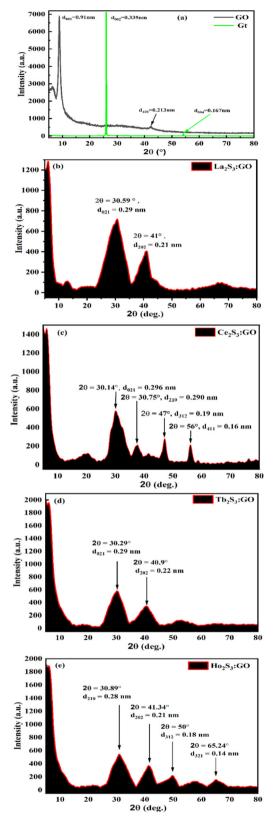


Fig. 8 XRD of (a) bare GO, (b)  $La_2S_3$ :GO, (c)  $Ce_2S_3$ :GO, (d)  $Tb_2S_3$ :GO, (e) Ho<sub>2</sub>S<sub>3</sub>:GO

Table 2 Elemental composition analysis for LSNR, GO, and LGT

Sr. no.	Name	C (%)	Н (%)	S (%)	Weight (mg)
1	GO	60.558	3.306	_	1.0752
2	La <sub>2</sub> S <sub>3</sub> :GO	18.782	2.669	2.134	1.0756
3	Ce <sub>2</sub> S <sub>3</sub> :GO	20.915	1.799	2.642	1.0995
4	Tb <sub>2</sub> S <sub>3</sub> :GO	19.264	2.480	1.764	1.0299
5	Ho <sub>2</sub> S <sub>3</sub> :GO	25.024	2.475	1.964	1.0295

Table 3 Elemental composition via EDS for LSNRs, GO, and LGT

System	Element	Weight (%)	Atomic weight (%)
Tb <sub>2</sub> S <sub>3</sub>	Tb	88.24	60.23
2 0	S	11.76	39.77
$Ho_2S_3$	Но	97.17	86.97
	S	2.83	13.03
GO	C	69.57	75.28
	O	30.43	24.72
Tb <sub>2</sub> S <sub>3</sub> -GO	С	55.90	64.10
2-3	O	40.42	34.65
	S	1.04	1.02
	Tb	2.64	0.23
Ho <sub>2</sub> S <sub>3</sub> -GO	C	56.19	63.84
2 0	O	41.23	36.01
	S	1.01	0.06
	Но	1.57	0.13

The specific characteristics of GO and the lattice are depicted with XRD. The specific features express the GO skeleton doped with LSNRs, as depicted with an XRD pattern for each LGT (Fig. 8). The XRD pattern of GO differs from that of the LGT which authentically indicates the doping mechanism due to the interaction and optimization activities of the FGs and 4f#e of LSNRs. Thereby, FT-IR, TGA, and XRD depict the structural lattice and features expressed as an energy spectrum denoted as DOS. These specific structural patterns supported by specific patterns of their morphology are depicted with SEM. The DLS data and distribution patterns of GO and the LGT in a most controlled pattern indicate the structural morphology and topography, which further synchronize their interacting PCP profiles. Structural illustrations focussing on almost each and every atom and their linkages in GO, LSNRs, and LGT indicate the stability of their structural frameworks despite subjecting them to photons. The elemental analysis further gives their structural constituents and their ratios to predict their presence in the structural lattice (Table 2). EDX spectroscopy was run on the samples, which elucidated their structural frameworks with their elemental composition (Table 3). Reports of our previous study with LSNRs and LGT of Pr<sub>2</sub>S<sub>3</sub>, Sm<sub>2</sub>S<sub>3</sub>, Eu<sub>2</sub>S<sub>3</sub>, and Gd<sub>2</sub>S<sub>3</sub> included XPS that confirmed the elemental composition and participation of each constituent stabilizing their resultant lattices.40

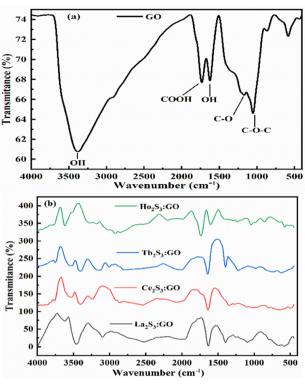
### 3.1. X-ray diffraction analysis

The single sharp GO peak at  $2\theta = 8.83^{\circ}$ , despite treatment with Gt, means a stronger oxidising mixture could not disrupt the

covalent bond between C atoms, but it weakened the van der Waals forces (VWF) binding the GO sheets. The GO matrix accommodated a lesser amount of LSNR dopant due to 4f #e in the doping<sup>43</sup> (Fig. 8a-e). The Gt plane at (002) is shifted to (001) in GO due to the intersheet distance (Fig. 8a, green line). LSNRs with LGT has reduced the intensity of minor peaks from 7000 for GO to 2200 for LGT8. The LGT generates a new intensity due to ROC of h<sup>+</sup> and e<sup>-</sup> holes resulting in lower intensity (Fig. 8a-e). XRD suggests there is no dissociation, with stronger interatomic forces generating prominent peaks at  $2\theta = 30.59^{\circ}$ and 41° for La<sub>2</sub>S<sub>3</sub>:GO with 0.29 and 0.21 nm d-spacings, respectively.<sup>27</sup> While Ce<sub>2</sub>S<sub>3</sub>:GO has peaks at  $2\theta = 20^{\circ}$ ,  $30.14^{\circ}$ , 30.75°, 47° and 57° with 0.443, 0.296, 0.290, 0.190 and 0.160 nm d-spacings of the lattice plane, respectively. 44 This has direct implications for PCPs, as a lattice plane develops different NHSs according to the d spacing. This indicates the relevance of PCPs to PCR and SCR, as the alignment of water affects electronic transitions. Tb<sub>2</sub>S<sub>3</sub>:GO with five half-filled electrons from the valence band (VB) to the conduction band (CB) causes the maximum number of collisions, unlike Ce<sub>2</sub>S<sub>3</sub>:GO, which generates sharper peaks at  $2\theta = 30.29^{\circ}$  and  $40.9^{\circ}$  with d-spacings of 0.29 and 0.22 nm.  $Ho_2S_3$ :GO has peaks at  $2\theta$  = 30.89°, 41.34°, 50° and 65.24° with 0.28, 0.21, 0.18 and 0.14 nm d-spacings, respectively, with 4f <sup>11e</sup> with four CBs that split into multiple peaks after with robust collisions, while the GO sheet remains structurally intact. LGT aligned with different peaks as GO is aligned at  $2\theta = 8.83^{\circ}$  with 6900 a.u. The FG synchronizes with a continuous electronic plane to interface the  $4f^{\#e} \sim FE$ collisions. The electronic configurations with higher d spacings for GO suggest stronger reducing activities as the FG might have pushed the LSNRs into the GO intersheet due to a higher mean free path (Fig. 8).

#### 3.2. FT-IR analysis

The bands at 3375, 1726, 1167 1051 and 1625 cm<sup>-1</sup> for the stretching of -OH, -COOH-C=O, C-O-C and OH, respectively, come GO alone (Fig. 9a). These FGs are aligned with different intensities around the various paths developing different lattices (Fig. 8). Their geometries influence the shear and surface energy for a PCP sensor. The bending vibrations of -OH, epoxide (-O-), and the skeletal rings of GO occur at 1624 cm<sup>-1</sup> with a small stretching vibration hump at 1372 cm<sup>-1</sup> from tertiary C-OH. The PCPs have repeated their state functions for conversion into the resulting product. The LGT has aligned the GO sheet to quench undesirable  $\pi \to \pi^*$  transitions of FGs and residual potential charges. FE exists with Ce2S3:GO and d-spacing with the restrictions, unlike Tb<sub>2</sub>S<sub>3</sub>:GO and Ho<sub>2</sub>S<sub>3</sub>:GO with 4f <sup>9e</sup> and 4f 11e, respectively. A new band at 1633 cm 1 indicates LSNR:GO optimization by FGs and 4f #e transitions. The stretching depends on electron-electron repulsion of 4f #e with nuclear charge.<sup>36</sup> The Born-Oppenheimer approximation optimizes LSNRs via  $4f^{\text{#e}} \sim \text{FE}$  clouds on the metal surfaces. The  $S^{2-}$  of LGT might not have stretched except for a minor simulation, as stretching from 2000 to 3000 cm<sup>-1</sup> indicates the C-C, C-H, C=C, and aromatic ring of LGT (Fig. 9). The stable GO sheet with prominent peaks decreases the stretching intensity. Multiple



FT-IR spectra analysis: (a) GO, (b) LGT

splits in stretching indicate the collision of 4f #e on interacting with an electronic cloud of FE (Fig. 9). The stretching of an FG with milder peaks within 500-1000 cm<sup>-1</sup> for La<sub>2</sub>S<sub>3</sub>:GO, Ce<sub>2</sub>S<sub>3</sub>:GO, Tb<sub>2</sub>S<sub>3</sub>:GO and Ho<sub>2</sub>S<sub>3</sub>:GO indicates GO doping with LSNR dopants (Fig. S2.9, ESI†). C-C is stabilized as FGs attached to the C atom are aligned to accommodate the dopants.40

# 3.3. C, H, N/S analysis

The structural changes are supported by the spatial arrangement of the constituent elements in the highest reproducible primitive lattice as a uniform wavevector of C, H, and S detected with a C, H, N, and S analyser, while the constituents of LSNRs and LGT are detected with elemental diffraction spectroscopy (Tables 2 and 3).

# 3.4. FESEM with EDS analysis

The morphologies of GO, LSNRs, and LGT have already been explained with SEM (Fig. 14). There is a further explicit focus on morphology with FESEM including elemental diffraction X-ray spectroscopy to explore their nucleating frameworks, elucidating their most primitive alignments (Fig. 10). Their morphologies elucidated their surface profiles at 1  $\mu$ m, 10  $\mu$ m, and 100 nm for GO, LSNRs, and LGT. The LSNR morphology suggests workable structural frameworks that constitute the specific atomic bonding and the sheet arrangements (Fig. 10B and D). GO constitutes the most symmetric and synchronized 2D sheet, closely interfaced with moderate VWF or Lennard Jones potential, as per the Prigogine intrinsic entropies and the Born-Oppenheimer approximation. The morphology which was further elucidated

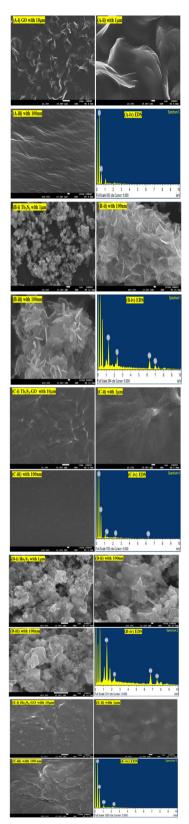


Fig. 10 The morphologies of GO, LSNRs and LGT obtained with an X-ray spectrometer (A, i-iv) GO, (B, i-iv) Tb<sub>2</sub>S<sub>3</sub>, (C, i-iv) Tb<sub>2</sub>S<sub>3</sub>:GO, (D, i-iv) Ho<sub>2</sub>S<sub>3</sub>, (E, i-iv) Ho<sub>2</sub>S<sub>3</sub>:GO.

with a 1 µm highly expanded view of the 2D GO sheet could be defined as top-to-bottom and bottom-up with a few areas and edges sharply diffracting laser light. Hence, this is a bottom-up tentropic surface with comparatively localised and ordered surface entropies within the closest interlink to resonate their energies for mutual transfer with single-valued  $\psi$ , enabling a PCR. The front view of the 1 µm scale morphology further indicates synchronized 2D GO sheets with the most symmetric upridges noted as basement spaced 2D GO sheets able to sense each minimum temperature and pressure fluctuation. These have been highly resolved at a scale of 100 nm; however, the longer 2D GO sheet with a structure similar to a carbon-fibre-like structure also reflects laser light (Fig. 10A). One side of the 2D GO fibrous sheets are anchored or adhered to a longer with few layered sheets. These sheets are highly and constantly fixed with their C atoms obtaining the 100% hybridization of potential energy *via*  $|\psi_{c-1}|$ ,  $|\psi_{c-2}|$ ,  $|\psi_{c-3}| \dots |\psi_{c-n}| = 100\%$  hybridization. The index n indicates the number of C atoms. Hence, 2D GO fibrous sheets slightly differ in an equipartitioning of energy, where their residual energies seem to facilitate a well-defined ordered tentropy with explicit friccohesity chemistry by balancing the PE of the basement sheet and the KE of fibrous sheets developing a remarkable interfacial  $\varphi$  to equilibrate the  $\varphi_{\rm Fibrous}$   $\times$  $\phi_{\mathrm{Basement}}$  matrix to balance a higher stability with definite FEs and hexagonal sheets. Therefore,  $|\varphi_{\text{Fibrous}}| \approx |\varphi_{\text{Basement}}|$  develops an energetic way of sticking to the higher surface area that could act as an anticoagulant, obeying autocatalytic friccohesity interfacial restrictions.

The morphology of the GO differs from LSNRs, as LSNRs could not act as 2D sheets, but rather act as one-dimensional entities with more residual energies at the terminal 3S<sup>2-</sup> and 2La<sup>3+</sup>. These spatial DOS could align them in a 3D energetic expression of LSNRs due to residual or electron-rich and charged defect-rich domains (Fig. 4a<sup>1</sup>). These defect-rich domains occupy the spatial dimension with  $\psi_{S^{2-}} \approx \psi_{2La^{3+}}$ within the energy clouds of S2- and La3+. These develop a cloud-like DOS with specific virtual wavevectors where the real nanorod alignments of these LSNR entities stabilize them (Fig. 10B and D) expressed as the Schrödinger equation:

$$E_{\psi} = [p^2/2m + V(x, t)]\psi \tag{1}$$

E is total energy of  $4f^{\#e}$  and  $\psi$  is the wavefunction, p is momentum, m is mass, and V is the potential energy (PE) at position x at time t. The position x at time t remains synchronized to symmetrize the alignments of the LSNRs. Positions  $x_1$ ,  $x_2, x_3, \dots, x_n$  align symmetrically with virtual boundaries. Hence,  ${
m Tb}^{3+}$  with  $4{
m f}^{8e}$  probably strengthens the bonding with  ${
m S}^{2-}$  (S<sub>16</sub> =  $1s^2$ ,  $2s^2$ ,  $2p^6$ ,  $3s^2$ ,  $3p^4$ ) where  $3p^4$  has UNPE  $3p_x^2$ ,  $3p_y^1$ ,  $3p_z^1$ . Hence, the electronic activities of 3p4e (S) and 4f8e (Tb) lead to the development of very functional interfacial adherence with highly optimized energies from 10  $\mu m$  and 100 nm (Fig. 10B). The most novel alignment of the  $\psi$  is  $\psi_n = \sqrt{a/2}\sin(n\pi x/a)$ , where the  $\psi$  could have  $\pi/2$ ,  $\pi$ , and  $2\pi$  coordinates of the wavevector. Hence, these hardcore wave mechanics laid down a function of interfacial alignment of the LSNRs and 2D GO

sheets (Fig. 10). Therefore, the LGT attained a matrix with the most QDs, highly integrated with the uniformly distributed lattice of the LSNR  $\approx$  GO energy plane without disrupting the basic structural framework of the LSNRs and GO. In fact, at 100 nm scale highly uniform energetic ODs are seen without energy spreading. Hence, the  $\psi_{LGT} \approx [\partial p^2/2m \pm V(x, t)]\psi$ integrate with the most synchronized arrangement depicted with the sharpest single light order.

These are compared with Ho<sub>2</sub>S<sub>3</sub>:GO having 4f <sup>10e</sup> without OD rates, where the most symmetric sin waves are depicted (Fig. 10E). Hence, the electronic cloud of 4f 10e influences the QDs of GO. The elemental compositions of GO, LSNRs and LGT are included in Table 3.

#### 3.5. TGA, DTG and DTA/DSC analysis

Ln<sub>2</sub>S<sub>3</sub>:GO responds to a rising temperature due to oscillation of  $4f^{\#e} \sim FE$  interfaces: see the overlapped DTG and TGA curves of LGT with weight (wt) loss in three steps (Fig. 11a-d).  $La_2S_3$ :GO decomposes at 100, 221 and < 800 with 20.77, 15.96 and 63.27% wt losses due to moisture, and single-bonded and double-bonded S<sup>2-</sup> with LGT surfaces, respectively. A maximum decomposition rate at 204 with a sharp exothermic peak predicts that La<sub>2</sub>S<sub>3</sub> will interact strongly with the FEs of GO. The  $\text{La}_2\text{S}_3$ :GO and  $\text{e}^-\text{h}^+$  interfacial energy optimized as  $q^+q^-/4\pi\epsilon r^2$ weakens  $\Delta H$  release. Ce<sub>2</sub>S<sub>3</sub>:GO at 215 and 717 lose maximum weight with sharp exothermic and endothermic peaks as 4f 1e interacts with FE, causing higher stability than GO (Fig. 11a). There are 27.4, 24.74 and 47.86% wt losses of Ce<sub>2</sub>S<sub>3</sub>:GO at 138, 246 and <800, respectively (Fig. 11c). Tb<sub>2</sub>S<sub>3</sub>:GO with 20, 31.46 and 48.54% wt losses at 119, 235 and <800 has sharp peaks with maximum decomposition rate at 92 and 212, respectively. Synchronized LSNRs and GO of LGT with a few new weight losses indicate their interfacial activities. The 4f 9e synchronized electronic cloud of Tb<sub>2</sub>S<sub>3</sub>:GO and Ho<sub>2</sub>S<sub>3</sub>:GO with 4f <sup>11e</sup> as a single crystal equalizes the PE to KE. Two sharp exothermic peaks indicate maximum energy release on interaction with GO due to its aligned FGs.

The 4f <sup>9e</sup> of Tb<sub>2</sub>S<sub>3</sub> with GO enhanced LGT stability (Fig. 11d). The maximum decomposition of Ho<sub>2</sub>S<sub>3</sub>:GO at 190 with a sharp exothermic peak through 79.83 and 20.17% wt loss does not suggest a crystalline structure. Ho<sub>2</sub>S<sub>3</sub>:GO shows a sharp weight loss at 198.19, as Ho3+ strongly aligns H2O dipoles as well as FE interfaces. Therefore, Ho<sub>2</sub>S<sub>3</sub>:GO reduces dyes in a shorter time by minimizing the collisions to increase the mean free path to overcome the QEB (Fig. 11e). The 4f#e enhance the thermal stability of both LSNRs and LGT by reducing the PCR time due to having a similar FE to GO (Fig. S1.0 and S1.1, ESI†). The difference in weight loss between LSNRs and LGT suggests stability due to doping. LSNRs alone lose weight in two steps due to the two strong interfacial forces of 2Ln<sup>3+</sup> and 3S<sup>2-</sup>. One is under the control of 4f 11e of Ln and the other under the control of S<sup>2-</sup> compared to LGT with thermal stability enhancing PCR and SCR activities. The LGT loses weight in three steps due to the interfacial domains of LSNRs and GO including their electronic clouds. GO which has a C-C lattice with equal heat capacities is disrupted at 193 with 82% wt loss with a sharp

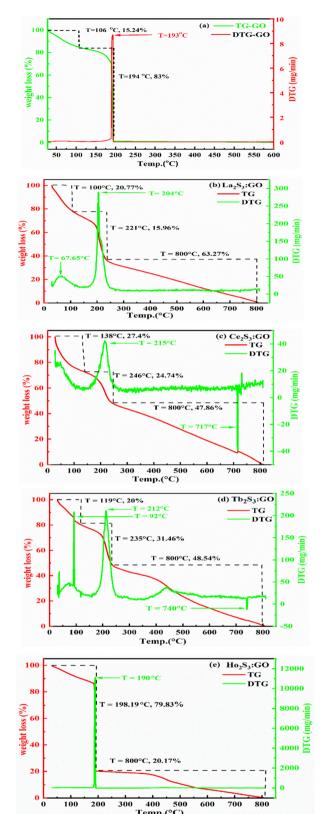


Fig. 11 TGA and DTG of GO and LGT.

peak. LSNR decomposition suggests the role of  $4f^{\#e}$  by aligning the lattices of  $La_2S_3$ ,  $Ce_2S_3$ , and  $Ho_2S_3$  but not  $Tb_2S_3$  which has a minor simulated weight loss with increasing the  $4f^{\#e}$  (Fig. S1.0 and S1.1, ESI $\dagger$ ) as:

$$T = \frac{\left(v + \frac{1}{2}\right)h}{2\pi} \sqrt{\frac{k}{\mu}}, \ \mu = \frac{m_1 m_2}{m_1 + m_2}$$
 (2)

here  $v \propto T$ ,  $T \propto \frac{1}{\sqrt{\mu}}$  and  $T \propto \sqrt{k}$ , and frequency depends on temperature and oscillations as per Hooke's law, and for the bond constant k,  $T \propto \sqrt{k}$ .

The basic structural units of LSNRs and LGT have been expressed on by subjecting them to the XRD and TGA, as their electronic DOS respond to the laser as per according to their wavevectors. The weight of a well-defined lattice also responds to the thermal energy, which induces the oscillation to transfer the oscillating energies via FRET with a 1st first-order weight loss transfer rate. Thus, the TGA and DTA also define the energy changes based on a the nature of the electron-electron collision (EEC) and electron-electron repulsion (EER) that interfere with the development in the same  $\varphi$  in the same  $\varphi$  (Fig. 12, green line). Thermal energy excites the unpaired electron within the wavevector with the patterns of the DOS of the DTA peak with LSNRs with comparatively sharper apex DTA along with differential heat flow, -7.0 to 10, -0.4 to 14, -7.0 to 12, and -0.4 to 13  $\mu$ V for La<sub>2</sub>S<sub>3</sub>, Ce<sub>2</sub>S<sub>3</sub>, Tb<sub>2</sub>S<sub>3</sub> and Ho<sub>2</sub>S<sub>3</sub>, respectively, at 400 °C apex temperature (Fig. S3.5, ESI†). Thus, the lattice of the LSNRs that responded at 400 °C remained fairly much the same that responded at 400 °C. However, the sharper DTA apex for an LGT is broader without any sharper peak-like shoulder. The DTA apex is broader than that of the LGT, which proves that the LSNRs have been fairly doped with into the 2D GO 2D sheet. These structural changes have fastened an energy flow within the LGT lattice on receiving the  $h\nu$  without causing the EEC and or EER by avoiding the recombination of the holes. Therefore, no the lack of a transitional peak in the DTA of LSNRs, GO, and LGT have inferred indicates smaller heat energy flow as equipartitioning the energy is equipartitioned (Fig. 12, green line). Initially LGT gains heat energy to get begin to oscillated and become ordered, where almost all the lattice points acquire a relaxation time. The unit becomes ordered within the coordinates of the initial state except the excited electronic state or the DOS. There is heat energy transfer without a loss and the PCR done is enacted as the gained heat excites an electron with a similar pattern where the  $h\nu$  could also push the electron within the a similar framework with to  $\psi_{\text{Nor,T}} \rightarrow \psi_{\text{Ref,T}}$ . The DTA analysis for LSNRs is included in the ESI† (Fig. S3.5).

### 3.6. Raman spectra analysis

The laser light interacts with the molecular vibrations in GO as well in LSNR:GO that inferred from two sharper peaks in the same range of Raman shift. The ratio of D and G bands infer indicates the participation of LSNRs in the GO sheet, and

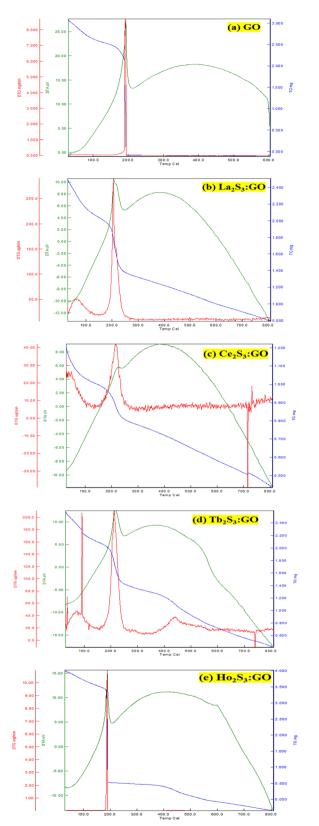
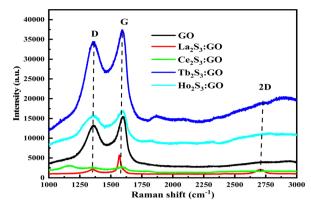


Fig. 12 Differential thermal analyses: (a) GO, (b-e) LGT.

creating disorder with the FGs and electrostatic moieties within the matrix. The  $I_{\rm D}/I_{\rm G}<1$  shows symmetric deformation on the GO sheet due to packing of LSNRs (Tables 1–3).



**Fig. 13** Raman spectroscopy with D, G and 2D bands for GO and LSNR@GO.

Table 4 Raman shifts with D and G, bands with their ratio for GO and LGT

LSNR@GO	D-shift (cm <sup>-1</sup> )	G-shift (cm <sup>-1</sup> )	$I_{\rm D}/I_{\rm G}$
GO	13 180	15 152	0.87
La <sub>2</sub> S <sub>3</sub> @GO	1827	5400	0.33
Ce <sub>2</sub> S <sub>3</sub> @GO	2501	2609	0.95
Tb <sub>2</sub> S <sub>3</sub> @GO	34 021	37 340	0.91
Ho <sub>2</sub> S <sub>3</sub> @GO	15 465	16 812	0.92

The D, G and 2D band peaks at 1350, 1580, and 2690 cm<sup>-1</sup> respectively indicate sp<sup>2</sup>-sp<sup>3</sup> and C-C stretching, respectively, and the intensity ratio shows the disorder of the GO sheet. The modulated bump on the 2D band at 2728 cm<sup>-1</sup> is caused by minor reorientation, and the higher intensity of the G band indicates there are more defects in the GO sheets.<sup>5</sup> The shift in energy bands is caused by the direct interaction of molecules with laser light. However, for the LGT, Raman spectra and intensities indicate the lattice and energy transfer behaviour of the LGT photocatalysts during the PCR. The Raman spectra have excellently developed similar patterns of the defunct activity of the 2D GO sheet doped with LSNRs in a most ordered manner (Fig. 13), as the D and G bands for GO, La<sub>2</sub>S<sub>3</sub>:GO, Ce<sub>2</sub>S<sub>3</sub>:GO, Tb<sub>2</sub>S<sub>3</sub>:GO and Ho<sub>2</sub>S<sub>3</sub>:GO are developed at 1350, 1580 cm<sup>-1</sup> with variable intensities. Thus, the basic matrix remained the same with a well-defined wavevector. The fixed wavevector of a geometry with highly ordered entropy allows the increasing 4f #e to equilibrate with the FE of the GO, developing a single-valued  $\psi_{\mathrm{LGT}}$  with negligible EEC and EER, leading to an efficient  $\sim 100\%$  FRET to dye for the PCR of a dye (Table 4).

#### 3.7. Scanning electron microscopy (SEM)

Micrographs of GO indicate a lace-like morphology of a thin sheet of C atoms in 2D at 20 m resolution with a single distribution pattern due to synchronized FGs illustrated with PCPs that focus the alignments. A single sharp peak in the GO thermogram indicates a single crystal structure (Fig. 11a). The increasing 4f <sup>#e</sup> alter the topography and FE distribution in LGT depicted with the patterns of XRD, FT-IR, TGA/DTG and DLS (Fig. 8, 9 and 15). The La<sub>2</sub>S<sub>3</sub>:GO fragments have reoriented due to VWF to hold La<sub>2</sub>S<sub>3</sub> within sheets, creating intersheet

interfaces. The LGT fragments of smaller sizes enhance surface charges with higher increasing rate of PCR and SCR compared to LSNRs and GO alone (Fig. 14a and b). Ce<sub>2</sub>S<sub>3</sub> with 4f<sup>1</sup> having a

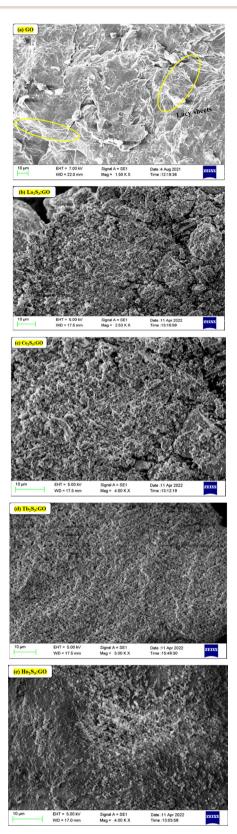


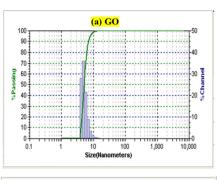
Fig. 14 SEM for surface morphology: (a) GO and (b-e) LSNR:GO.

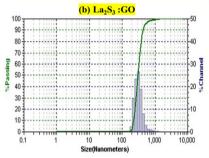
maximum number of CB reoriented the GO sheets with the  $\psi$  in different  $\varphi$  and the reorienting energies spatially aligned the sheets. The upgraded ridges and furrows indicate interconnected nanorods (Fig.  $4a^1$ ).  $Tb_2S_3$  with  $4f^9$  having  $\psi$  with a highly intensified RE unlike  $\psi_1 \iint \psi_2 \iint \psi_3 \iint$  unpaired electronic states with  $[\psi_1 \approx \psi_2 \approx \psi_3]^2$  probability. Subscripts 1, 2, 3 on  $\psi$  denote electron transitions from 1  $\rightarrow$  2  $\rightarrow$  3 energy states of the 4f<sup>#e</sup> suborbital. Critical analysis of the micrographs of GO, La<sub>2</sub>S<sub>3</sub>:GO, Ce<sub>2</sub>S<sub>3</sub>:GO, Tb<sub>2</sub>S<sub>3</sub>:GO and Ho<sub>2</sub>S<sub>3</sub>:GO indicate topographical differences at 4f #e (Fig. 14). The changes are caused by the splitting energy of Ln<sub>2</sub>S<sub>3</sub>:GO with GO due to 4f #e. LSNR dopants have influenced the GO morphology due to  $4f^{\#e} \sim FE$  on aligning a GO matrix vis-à-vis LGT. La<sub>2</sub>S<sub>3</sub>:GO, Tb<sub>2</sub>S<sub>3</sub>:GO, and Ho<sub>2</sub>S<sub>3</sub>:GO produced almost equally distributed energy patterns, unlike Ce<sub>2</sub>S<sub>3</sub>:GO. Equal distribution of similar lacings of GO sheets are split into equal distributions as each GO sheet participates in LGT formation. The unpaired electron of Tb<sup>3+</sup> symmetrises a GO topography, accepting more hv for PCR and SCR with Tb<sub>2</sub>S<sub>3</sub>:GO (Fig. 14d). Ho<sub>2</sub>S<sub>3</sub> with 4f<sup>11e</sup> develops  $\psi_{4f11}$  with the least QEB, inducing a symmetric topography RE of FE. The  $\psi_{4\text{f}11}$  and  $\psi_{\text{FE}}$  in the same  $\varphi$  have developed a single crystal with a sharp peak, as depicted with TDA/DTG (Fig. 11e).

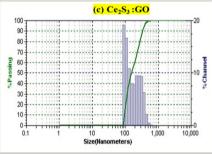
Tb<sub>2</sub>S<sub>3</sub>:GO and Ho<sub>2</sub>S<sub>3</sub>:GO as robust photocatalysts and adsorbents receive maximum  $h\nu$  almost in the same  $\varphi$  as the GO sheets for uninterrupted PCR and SCR.

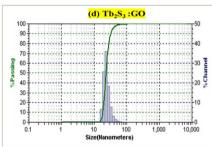
#### Dynamic light scattering (DLS) 3.8.

The  $\zeta > 0$  of GO indicates aligned h<sup>+</sup> holes on a surface of LGT and LSNRs either from  $S^{2-}$  or  $Ln^{3+}$  charges. LGTs receive  $h\nu$ with different capacities. A broader La<sub>2</sub>S<sub>3</sub> distribution pattern with GO sheets indicates weaker 4f 0e interactions with FEs. The oxidation potential of La3+ and reduction potential of S2- are unable to generate a caging of water (Fig. 15). La<sub>2</sub>S<sub>3</sub> is actively distributed with less passage while on doping with GO there is maximum capping and a decrease in the passage of population on interaction of LSNRs and FEs. Ce2S3 with 4f1e develops prominent collisions with a stoichiometric ratio of GO and LSNRs for LGT with a  $4f^{0e} \rightarrow 4f^{11e}$  interface The conductivities of LSNRs and LGT are in the range 17 to 43  $\mu S \; cm^{-1} \; compared$ to the 50  $\mu S$  cm<sup>-1</sup> of GO alone, so the 4f<sup>#e</sup> and FE interactions might have occupied the surface charges and mildly responded to an electrical potential with a lower mobility (Table 5). The distribution of La2S3 is scattered while that of La2S3:GO is narrow, as GO has an ordered lattice due to similar GO sheets. There is a narrower range of distribution between 100 and 1000 nm at -0.49 mV compared to that at 22.24 mV, as the La<sup>3+</sup> develops Coulombic interaction with negative holes that confine C atoms (Fig. S3.4, ESI†). The redistributing electronic clouds equipartition the electronic energy of S<sup>2-</sup> to interact with the h<sup>+</sup> holes or electron-deficient sites of GO. Their interactions have narrowed down the distribution range of La<sub>2</sub>S<sub>3</sub>:GO (Fig. 14). These changes are supplemented by the TGA of GO and LGT, with major structural disruption of La<sub>2</sub>S<sub>3</sub> at 365 and of GO at 194, but La<sub>2</sub>S<sub>3</sub>:GO has elevated the GO stability of La<sub>2</sub>S<sub>3</sub>:GO with a major structural disruption at 221 (Fig. 11). These disruptive









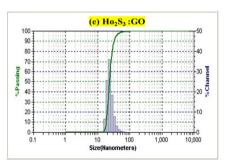


Fig. 15 Size (nm) of (a) GO and (b-e) LGT analysed by DLS

activities indicate the doping mechanism and pattern of GO with La<sub>2</sub>S<sub>3</sub>. The 3e<sup>-</sup> + La<sup>3+</sup> local -C-C-C- network is

simultaneously accompanied by 3h<sup>+</sup> + 3S<sup>2-</sup>. The interfaces of La<sub>2</sub>S<sub>3</sub> and GO narrow down the distribution range with an active state due to FRET. A similar pattern is noticed with Tb<sub>2</sub>S<sub>3</sub> and Tb<sub>2</sub>S<sub>3</sub>:GO, as their distribution range remains almost the same with a sharp lattice disruption at 235 (Fig. 1d). Thus, the Tb<sup>3+</sup>-3e<sup>-</sup> and S<sup>2-</sup>-2h<sup>+</sup> electronic strings are influenced by 4f<sup>9e</sup> due to their oscillations. However, the Ln<sup>3+</sup>-3e<sup>-</sup> and S<sup>2-</sup>-2h<sup>+</sup> loops are quantized and their localized quantum dots may be equally distributed in Ln<sub>2</sub>S<sub>3</sub> and Ln<sub>2</sub>S<sub>3</sub>:GO. The FGs of GO could opt to interact with Ln3+-3e and S2--2h electronic strings as Ln<sub>2</sub>S<sub>3</sub>:GO. Surprisingly Ce<sub>2</sub>S<sub>3</sub> and Ho<sub>2</sub>S<sub>3</sub> with their templates produced a fine distribution in the 10 to 1000 nm range. Ce<sup>3+</sup>-3e<sup>-</sup> and Ho<sup>3+</sup>-3e<sup>-</sup> with S<sup>2-</sup>-2h<sup>+</sup> are simulated with 4f<sup>1e</sup> and 4f<sup>11e</sup> of Ce<sup>3+</sup> and Ho<sup>3+</sup>, respectively, which are duly supplemented with the TGA pattern (Fig. 11c and e). Since these electronic strings substantially influence their solubilizing activities, so their PCPs indicate the thermodynamic state function. The Ln<sup>3+</sup>-3e<sup>-</sup>(GO) and S<sup>2-</sup>-2h<sup>+</sup>(GO) interfaces are highly functional and localized cohesively as per their respective charges. So, solvent adherence could cause further alignment or reorientation despite their cohesiveness, and their friccohesity index could reveal a unique interaction mechanism. Ln<sub>2</sub>S<sub>3</sub> and Ln<sub>2</sub>S<sub>3</sub>:GO have developed new interaction centres unlike the others. Their  $h\nu$ receiving and phonon generating activities have been found to be directly interconnected with their PCPs. Therefore, the photocatalysts were solubilized for PCP, PCR, and SCR studies. Their electronic interactions could be influenced by sonication to induce some electronic arrays in contact with 2Ln<sup>3+</sup>-3S<sup>2-</sup> as S = Ln-S-Ln = S nanorods. Thus, the PCPs and DLS profile elucidate applications to reduce and recycle wastewater. Ho<sub>2</sub>S<sub>3</sub> has shown a 5.35 PDI which was reduced to 1.369 with Ho<sub>2</sub>S<sub>3</sub>:GO, as the GO capped the Ho<sub>2</sub>S<sub>3</sub> around its surface. These are understood with TGA, in which it has developed multiple disruptive temperatures, but on doping developed a single crystal lattice resulting in a sharper single line at 198 against the 194 of GO (Fig. 11a and e). Ho<sub>2</sub>S<sub>3</sub> has elevated the GO disruption temperature by 4. Thus, the PDI and lattice disruption via TGA could have a unique relationship. The diameter of La2S3 is 457 nm, the highest among the series, which is also explained by TGA due to the Gaussian distribution pattern of heat (Fig. 15 and Fig. S3.4, ESI†).

Except for Ce<sub>2</sub>S<sub>3</sub>:GO and Ho<sub>2</sub>S<sub>3</sub>:GO, their mobilities are almost in the same range. La<sub>2</sub>S<sub>3</sub>:GO and Tb<sub>2</sub>S<sub>3</sub>:GO have negligible  $\zeta$  values, which may be due to the negative holes of GO. Their conductivity remains within 50 μS cm<sup>-1</sup> and responds to electrode charges on subjection to the probe (Table 5). A detailed study of DLS for GO, LSNR and LGT is incorporated in Fig. S3.4 (ESI†).

# LSNR and LGT solutions for PCPs, PCR and SCR

The NHSs of LSNRs and LGT with BSA could have been deformed irreversibly on acquiring unequal activation energy due to the selective charge clouds of Ln3+-3e- and S2--2h+

(c) LSNR with BSA (b) LGT) (d) LGT with BSA

Aq-solutions (a) LSNR (b) LGT (c) LSNR-BSA (d) LGT-BSA

electronic strings with the LGT. The highly sensitive survismetry technique measures viscous flow time (vft) and pdn at 283.15 to 315.15 K.45 Lifting a sample from a reservoir to a buffer bulb under a reduced pressure within capillaries avoided the mechanical shock that could have disarrayed the FE. A sample was equilibrated in the buffer bulb for 3 min before allowing for a backward flow to a CPU (controlled pressure unit) with a hassle-free capillary flow in survismetry. 46 PCPs for 0.002-0.010% @0.004% of LSNRs, LGT alone and with 0.010% aq-BSA at 298.15 K and 0.010% with Milli-Q water at 283.15 to 315.15 K indicate a monodispersion and structural alignment of the solution (Fig. 16a-d). LSNRs and LGT alone did not aggregate but developed a gel with BSA (Fig. 16c and d). Turbidity increases with increasing rate of g% of LSNR and LGT as 4f #e and FEs could align slightly, holding water molecules haphazardly. Their PCPs and thermodynamic parameters depict the solubilizing activity and interactions of aq-BSA with LSNRs and LGT. Viscosity elucidates the effect of Ln<sup>3+</sup>-3e<sup>-</sup> and S<sup>2</sup>-2h<sup>+</sup> in bulk while there is a surface tension on the surfaces. These have been used as the functional interfaces to explore the activities of LSNRs and LGT for widening their applications in areas of PCR and SCR. La<sub>2</sub>S<sub>3</sub> $\Delta H \ll 0$  for La<sub>2</sub>S<sub>3</sub>, and  $\Delta H < 0$  for Ce<sub>2</sub>S<sub>3</sub>, Tb<sub>2</sub>S<sub>3</sub>, and Ho<sub>2</sub>S<sub>3</sub> indicate a milder disruption of the water structure and LSNRs and LGT could have been expeditiously realigned on releasing H. Ln<sup>3+</sup>-3e<sup>-</sup> and S<sup>2-</sup>-2h<sup>+</sup> seem to develop stable dipoles that are explained by their photocatalyzing activities. LSNRs and LGT as a single unit with expeditious molecular motions disrupted hydrogen bonding (HB) on decreasing H and increasing S, as supported by DLS (Table 5 and Tables S1.4-S2.1, ESI†).

# 4.1. Density ( $\rho$ , g cm<sup>-3</sup>)

The  $\rho$  of LSNRs and LGT alone and with aq-BSA indicate interacting activities as a function of temperature and concentration. The  $\rho$  values are in the order of aq-BSA > La<sub>2</sub>S<sub>3</sub>:GO >  $La_2S_3$ :GO-BSA >  $La_2S_3$ -BSA >  $La_2S_3$  at 288.15, 298.15 and 310.15 K, predicting stronger aq-BSA and weaker aq-Ln<sub>2</sub>S<sub>3</sub> interactions. The electrostatic salt bridges of BSA strongly

attract the H<sub>2</sub>O dipoles, including Ln<sup>3+</sup>-3e<sup>-</sup> and S<sup>2-</sup>-2h<sup>+</sup> electronic strings. Thus, La<sub>2</sub>S<sub>3</sub> and La<sub>2</sub>S<sub>3</sub>:GO with aq-BSA decreased  $\rho$  by 0.039% and 0.009%, 0.039% and 0.0095%, and 0.048% and 0.16% at 288.15, 298.15 and 310.15 K, respectively as the BSA was weakly reoriented to optimize their interacting surfaces. The tryptophan and peptide bonds (PB) of BSA strongly interacted with water and Ce2S3:GO by occupying a majority of FEs that salted out water dipoles. BSA decreased the ρ of aq-Ce<sub>2</sub>S<sub>2</sub> and aq-Ce<sub>2</sub>S<sub>3</sub>:GO by 0.0027% and 0.0090% at 288.15 K and 0.0026 and 0.0040% at 298.15 K, unlike the increases of 0.0029% and decreases of 0.034% at 310.15 K, respectively. Ce<sub>2</sub>S<sub>3</sub>:GO with increasing temperature could have generated vibrant oscillations, weakening BSA interactions with H<sub>2</sub>O and could attract GO. BSA decreased the  $\rho$  with Tb<sub>2</sub>S<sub>3</sub> and Tb<sub>2</sub>S<sub>3</sub>:GO by 0.006% and 0.011%, 0.007% and 0.014%, and 0.013% and 0.039% at 288.15, 298.15 and 310.15 K, respectively, as the H2O accommodated them on holding BSA as well as  $Tb_2S_3$  and  $Tb_2S_3$ :GO. BSA decreased the  $\rho$ of Ho<sub>2</sub>S<sub>3</sub> and Ho<sub>2</sub>S<sub>3</sub>:GO by 0.0053% and 0.0131%, 0.0073% and 0.0160%, and 0.0073% and 0.080% at 288.15, 298.15 and 310.15 K, respectively (Tables S8.1 and S8.2, ESI†). Thus, the H<sub>2</sub>O, which was associated with BSA, Ln<sub>2</sub>S<sub>3</sub> and Ln<sub>2</sub>S<sub>3</sub>:GO was realigned with weaker force but adequate adherence. BSA developed stronger interactions with H2O dipoles associated with LSNR:GO and LSNRs, with stronger internal pressure of FEs (Fig. 16). The H<sub>2</sub>O dipoles interact with themselves and the salt bridge, and the electrostatics of BSA could have normalized Ln<sub>2</sub>S<sub>3</sub> and Ln<sub>2</sub>S<sub>3</sub>:GO. GO engages the LSNRs in LSNR:GO via weakly interacting H<sub>2</sub>O dipoles. The covalently bonded S<sup>2-</sup> and Ln<sup>3+</sup> of LSNRs and LGT develop NHSs, as NH<sub>3</sub><sup>+</sup>, COO<sup>-</sup>, >C=O, and PB along with tryptophan develop stronger hydrophilic-hydrophilic interactions (HHI), aligning LSNR:GO expeditiously during a PCR.31

#### 4.2. Viscosity (η, mPa s)

The  $\eta$  indicating the vft of Ln<sup>3+</sup>-3e<sup>-</sup> and S<sup>2-</sup>-2h<sup>+</sup> aligned in aqueous solution is calculated with eqn (3) (Tables 6, 7 and Table S1.2, ESI†).

$$\eta = \left(\frac{t}{t^0}\right) \left(\frac{\rho}{\rho^0}\right) \eta_0, \quad \eta_r = \left(\frac{\eta}{\eta_0}\right)_T (1.0) \text{ and } h_r = 1 + 2.5\phi \quad (3)$$

where  $\rho$  is the density of solution,  $\rho^0$  is the density of water,  $\eta_0$ is the viscosity of water,  $\eta_r$  is the relative viscosity, and vft is an average relaxation time ( $\tau$ ). Decreasing viscosities of LSNRs and LGT alone and with BSA indicate weaker interactions due to competitive adherence with S = Ln-S-Ln = S, and aq-GO, and their interfacial charges reorient and align the NHSs, generating cohesive activities (Tables 6, 7 and Fig. 17). The  $\eta$  values in the order  $La_2S_3 > Ce_2S_3 > Tb_2S_3 > Ho_2S_3$  with increasing the 4f #e and temperature indicate that electron spin via electronelectron collisions weakens H2O adherence, promoting coherence. The stronger interaction of GO with Ho<sub>2</sub>S<sub>3</sub> engages almost all the FEs, with stronger  $H_2O$  adherence increasing  $\eta$ values. The  $\eta$  values for LGT decreased on the GO and LSNRs mutually interacting via their respective NHSs. The NHSs align, generating mutual cohesive forces with sharply weakening adherence of H2O, as cohesive activities induce continuity,

equilibrating the adhered H<sub>2</sub>O as bulk H<sub>2</sub>O. Thereby monolayer adherence of H<sub>2</sub>O with comparatively weaker coherence

Ag-LSNR (a) La,S Ce<sub>2</sub>S<sub>3</sub> Tb<sub>2</sub>S<sub>3</sub> η (mPa.S) Ho<sub>2</sub>S<sub>3</sub> 0.7 0.6 310 315 1.0 Aq-LGT **(b)** 0.9 0.8 Ce<sub>2</sub>S<sub>3</sub>-GO η (mPa.S Tb<sub>2</sub>S<sub>3</sub>-GO 0.7 Ho<sub>2</sub>S<sub>3</sub>-GO 0.6 285 315 290 300 305 310 Temp.(K) Aq-LSNR with BSA (c) 0.9 ار (mPa.S) 10.7 La<sub>2</sub>S<sub>3</sub>-BSA Ce<sub>2</sub>S<sub>3</sub>-BSA Tb<sub>2</sub>S<sub>3</sub>-BSA Ho<sub>2</sub>S<sub>3</sub>-BSA 0.6 295 300 Temp.(K) 305 310 285 290 315 Aq-LGT with BSA 0.9 a<sub>2</sub>S<sub>3</sub>-GO-BSA Ce<sub>2</sub>S<sub>3</sub>-GO-BSA Tb<sub>2</sub>S<sub>3</sub>-GO-BSA Ho2S3-GO-BSA 0.6 285 290 300 305 310 295

Fig. 17 Viscosity of aq-LSNR and aq-LGT with BSA at 5 T.

enhances the  $\eta$  at lower temperature. These activities sensitively visualised a change compared to an uneven magnitude of  $\gamma$ data at higher temperature (Fig. 17c and d). The GO controls a CAC of LSNRs, but BSA has disrupted the LSNRs aggregating as colloids with a lower value due to multiple NHSs tending towards a monolayer (Fig. 17b). BSA capping of LSNRs and LGT has weakened their cohesive forces, as BSA disrupts the aligned water dipoles, forming multiple NHSs with slight monodispersion (Fig. 17b and d). The values of La<sub>2</sub>S<sub>3</sub>, Ce<sub>2</sub>S<sub>3</sub>, and Tb<sub>2</sub>S<sub>3</sub> intersect at 0.006 g% unlike Ho<sub>2</sub>S<sub>3</sub> with stronger adherence of H<sub>2</sub>O. The values decreased with concentration of  $La_2S_3 \sim H_2O$ . Additionally,  $Tb_2S_3$  with aq-BSA increased the  $\eta$ values due to the half-filled electronic spins of 4f 9e causing oscillations aligning H2O dipoles for comparatively stronger adhesion. On increasing Ho<sub>2</sub>S<sub>3</sub>, g% with aq-BSA did not produce a higher  $\eta$  value unlike Tb<sub>2</sub>S<sub>3</sub> at 0.006 and 0.010 g% due to 4f 9e, as an unpaired electron does not align the H<sub>2</sub>O in light of the NHS of BSA (Tables S6.3 and S6.4, ESI†). The BSA monodispersion with increasing LSNR g% develops LSNR ~ PB interactions on solvent adherence (Fig. S1.2, ESI†). LSNRs align the surface charges of BSA to develop multiple NHSs, resulting in a weaker NHS on a continuously flickering H2O. The ordered trends for values for aq-LGT-BSA compared to aq-BSA alone and aq-LSNR-BSA redistributes and realigns the surface charges with concentration. The aq-BSA regularly adheres to LSNRs with increasing g% on engaging a solvent. Amounts of 0.002 g% for La<sub>2</sub>S<sub>3</sub> and Ce<sub>2</sub>S<sub>3</sub> produced almost the same values unlike the situation at 0.010 g%. The values of La<sub>2</sub>S<sub>3</sub>:GO-BSA, Ce<sub>2</sub>S<sub>3</sub>:GO-BSA, Tb<sub>2</sub>S<sub>3</sub>:GO-BSA and Ho<sub>2</sub>S<sub>3</sub>: GO-BSA depend on g% (Tables S6.3 and S6.4, ESI†). The trend in polynomial  $\eta$  for La<sub>2</sub>S<sub>3</sub>, Ce<sub>2</sub>S<sub>3</sub> and Tb<sub>2</sub>S<sub>3</sub> on increasing g% at constant temperature predicts saturation at 0.006 g% (Fig. S1.2a, ESI†). Aq-Ho<sub>2</sub>S<sub>3</sub> mildly increased the  $\eta$  values, unlike the case at 0.006 g% due to hydrated LSNRs as the Ho<sub>2</sub>S<sub>3</sub> nanorods could have aligned H<sub>2</sub>O dipoles (Fig. S1.2b, ESI†). Increasing Ho<sub>2</sub>S<sub>3</sub> g% has mildly realigned a saturated  $H_2O$  with stronger adherence, but decreases the  $\eta$  values due to energetically disrupting the aligned NHSs (Fig. 5b). The residual charges align the LSNR- $H_2O$  interface with  $G_{net}$  =  $G_{\rm H_2O-H_2O} \neq G_{\rm interfacial}$  as a polynomial relation. Thus, increasing the g\% of Ho<sub>2</sub>S<sub>3</sub> disrupts the NHS, which affects both the adhesive and cohesive force. Ho<sub>2</sub>S<sub>3</sub> does not disrupt the H<sub>2</sub>O and is salted out, requiring sonication to remain suspended. La<sub>2</sub>S<sub>3</sub>, Ce<sub>2</sub>S<sub>3</sub>, and Tb<sub>2</sub>S<sub>3</sub> decreased the  $\eta$  values by weakly disrupting the adhering NHS that distinguished the  $G_{\rm H_2O-H_2O}$  and  $G_{\rm interfacial}$  on sonication. LGT realigns  $\rm H_2O$  to enhance aq-LSNR solubility with higher  $\eta$  values due to effective adherence increasing  $\tau$ . The rheology of aq-Ln<sub>2</sub>S<sub>3</sub> with increasing temperature equalizes with their oscillation with certain  $\psi$  equipartition of the energy. The adherence of H<sub>2</sub>O to Ln<sub>2</sub>S<sub>3</sub> is equal, however, at lower temperature and the viscosity values differ due to different oscillations and  $\psi$ s. The shift from Ln<sub>2</sub>S<sub>3</sub> to Ln<sub>2</sub>S<sub>3</sub>:GO at 298.15 K, indicates GO doping reducing energy equipartition (Fig. 17). However, BSA with Ln<sub>2</sub>S<sub>3</sub> and Ln<sub>2</sub>S<sub>3</sub>:GO have indistinguishable viscosities due to an equal adherence pattern. Robust oscillations of electron

clouds of GO with LGT influence adherence via  $\psi_s$  (Fig. 17a and b). Interactions of Ln<sub>2</sub>S<sub>3</sub> and Ln<sub>2</sub>S<sub>3</sub>:GO with BSA via NHS have sandwiched Ln<sub>2</sub>S<sub>3</sub>:GO with BSA together as incompressible, that lowers the  $\eta$  with increasing temperature. The  $\eta$  of packed LSNR-BSA and LGT-BSA support expeditious resonating energy transfer for equipartition, where no energy-deficient sites are generated by varying the temperature.

Due to its salt bridges and electrostatic sites, BSA is closely packed in between the cages or NHSs of LSNRs and LGT, lowering the  $\eta$  values equally (Fig. 17c and d). Tb<sub>2</sub>S<sub>3</sub>-BSA and Ho<sub>2</sub>S<sub>3</sub>-BSA interactions are weakened at 0.006 and 0.010 g% compared to Tb<sub>2</sub>S<sub>3</sub> and Ho<sub>2</sub>S<sub>3</sub> alone, as 4f #e ~ PB linkages and stronger electrostatic sites of BSA affect NHS alignment (Fig. S1.2c and d, ESI†). The  $\eta$  values of BSA with LGT are increased on strengthening the FE of GO and electrostatic interactions of BSA that are more soluble with LGT than LSNRs, as no FE networking exists with the LSNRs on disruption of HB. The LGT-BSA biotemplate increased BSA solubility, due to expeditious charge equalization through 4f #e ~ FE interfaces *via* FRET increasing the  $\eta$  values (Fig. S1.2d, ESI†).

# 4.3. Surface tension $(\gamma, mN m^{-1})$

Disruption of H2O adhering to LSNRs and LGT alone and with BSA affect the cohesive force of the medium. The  $\gamma$  elaborates the structural interaction with the solvent and pdn indicates the cohesive force, and  $\gamma$  is calculated with pdn and density values (Tables 8, 9 and Table S1.1, ESI†) as:

$$\gamma = \left(\frac{n^0}{n}\right) \left(\frac{\rho}{\rho^0}\right) \gamma^0 \tag{4}$$

where  $\gamma^0$ ,  $n^0$ ,  $\rho^0$  and  $\gamma$ , n,  $\rho$  are the surface tension, pdn, and density of water and solution, respectively. The surface tension of aq-LGT with BSA is higher than aq-LSNR with BSA, aligning multiple NHSs with stronger cohesive forces (Tables 8, 9 and Fig. 18). LGT developed a critical aggregation temperature (CAT) at 298.15 K similar to a critical solution temperature. 47 The H<sub>2</sub>O dipoles cap LGT to transfer a resonating energy with a moderate VWF (Fig. 18b). The CAT is controlled by 4f #e, as GO sheets adhere to LSNRs via NHSs that equalise the surface energy (SE). The FE with SE of LSNRs exchange their oscillations. Aq-LSNR does not favour CAT; thus, the  $4f^{\#e} \sim FE$  works to develop multiple linkages to develop CAT at ~298.15 K (Fig. 18a and b). The GO wrapped the LGT and a CAT could not distinguish an SE while neither the LSNRs nor LGT produce a CAT with BSA. BSA increasing time binds the LSNRs via NHSs with different surface charges and heat capacities around LSNRs via cohad. On partially engaging the cohesive and adhesive forces expressed as cohad. BSA having higher SE and a lower  $\gamma$  with aq-LGT developed a critical micelle temperature (CMT) at 298.15 K (Fig. 18d). BSA did not initiate a CMT due to its PB salt bridges, and electrostatic dipoles that stabilize the SE of LGT at 298.15 K. The SEs of GO and BSA both mutually adhere at 298.15 K with the highly sensitive surfaces for expeditious energy transfer via VWF. The trend of change in  $\gamma$  values unlike the  $\eta$  values depicts an adherence of NHS around a core of LSNRs. The cohad infers a balances in the

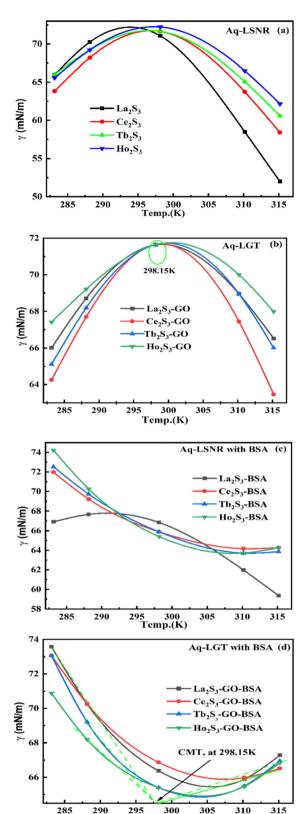


Fig. 18 Surface tension of aq-LSNR and aq-LGT with BSA at ST.

cohesive and adhesive forces of the solvent as well as the LSNR and LGT for the thermodynamic and kinetic stabilities of their **Energy Advances Paper** 

aqueous solutions preventing their coagulation (Fig. 18). The  $\eta$ ,  $\gamma$ , and  $\sigma$  indicating aq-Ho<sub>2</sub>S<sub>3</sub> with three unpaired electrons are more sensitive than other LSNRs due to the same contraction effect with stable dipoles (Fig. S1.3a, ESI†). Surfaces of LSNRs are capped with GO in manifold layers to attract the electrostatic centres of PB. LSNR:GO-BSA strengthens the electrostatic interactions with dipolar water. The lower γ values for aq-Tb<sub>2</sub>S<sub>3</sub> indicate stronger adherence with LGT and BSA with a higher (Fig. S1.3d, ESI†). The 4f #e engage the FE to get solvated or NHSs with increasing, and LSNR:GO with BSA further increases the adherence with increasing. The  $4f^{\#e} \sim FE$  photonic interfaces of LGT with BSA could store energy through aligned dipoles<sup>45</sup> by preventing the recombination of holes. The electrostatic linkages with  $4f^{\#e} \sim FE$  could develop the functional interface with a protein of specific charge with highly stabilized dipoles from 285 to 297 K; the  $\gamma$  value saturates but increasing H<sub>2</sub>O drastically develops an NHS.

### 4.4. Friccohesity ( $\sigma$ , S cm<sup>-1</sup>)

HB disruption aligns H2O around LSNRs and LGT separately and with BSA develops specific adherence with the cohesive force that supplements the respective FRET to  $E_a$ . It equalizes a stabilizing energy to develop  $\sigma$ , calculated from pdn and vft (Tables S1.1 and S1.2, ESI†):

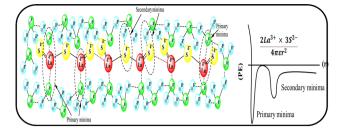
$$\sigma = \sigma^0 \left( \frac{t}{t_0} \pm \frac{B}{t} \right) \left( \frac{n}{n_0} \pm 0.0012(1 - \rho) \right),$$

$$\sigma = \sigma^0 \left( \frac{t}{t_0} \right) \left( \frac{n}{n_0} \right) \text{ and } \sigma = \frac{\sigma^0}{t_0 n_0} (t \times n)$$
(5)

where  $\sigma$  and  $\sigma_0$  are the friccohesity of solution and solvent, respectively,  $\pm B/t$  are kinetic and  $\pm 0.0012(1 - \rho)$  buoyancy corrections, other symbols are as usual. Corrections of the order of  $1 \times 10^{-7}$  are omitted. The common physicochemical parameters of solvents are rationalized as a Mansingh constant  $(M_c)$  that finds an interface between adhesive and cohesive forces.  $M_c$  depicts a mutual shear stress and the SE coefficient of a solvent for GO, LSNRs and LGT with BSA calculated with eqn (6):

$$M_{\rm c} = \frac{\sigma^0}{t_0 n_0} \tag{6}$$

 $M_c$  as a quantizing coordinate of aligned  $H_2O$  dipoles with LSNRs, LGT and BSA indicates homogenization with equipartition of their energies with a solvent via FRET. The  $M_c$  operator transforms cohesive to oscillating energies through a mutual electrostatic reorientation of solvent molecules around the LSNRs and LGT. The friccohesity of aq-LSNR, and LGT alone and with BSA indicate spontaneous solvent adherence (Tables 10, 11 and Fig. 20). The  $4f^{\#e} \sim FE$  interfaces of LSNRs, LGT and BSA are accessible to the internal salt bridges and electrostatic dipolar linkages of BSA to attract H<sub>2</sub>O dipoles. The H<sub>2</sub>O dipole interacts with each constituent of LSNRs, LGT and BSA via quantized negative and positive holes to adhere H<sub>2</sub>O. It avoids nanoclusters to promote PCR and SCR activities. Friccohesity generates several NHSs with different charges on surfaces that initiate the



Primary and secondary minima via 4f<sup>#e</sup> ~ H<sub>2</sub>O.

secondary global energy minima, stabilizing their NHSs for the required cohesive force. The friccohesity illustrates a mechanism and pattern of primary and secondary minima as the LSNRs, LGT and BSA interact differently through the NHS. Their NHSs mutually equalize the energy of a medium (Fig. 20).

Secondary minima attained via LSNR  $\sim$  HS, LGT  $\sim$  HS, and BSA ~ HS interactions facilitate a FRET for mutual homogenization. The friccohesity for partitioning the secondary quantization of cohesive forces was transformed into oscillatory energy by reducing the QEB of an NHS (Fig. 19). The  $\sigma$ represents the CMT of LSNRs and LGT with BSA at 295 and 302.5 K, respectively, aligning the FEs at 302.5 K compared to  $\eta$ . It depicts a higher resolution incorporating the role of each constituent for NHS quantization (Fig. 20a and d). The  $\sigma$  values illustrate the reengineering of interactions to analyse a contribution from each charge to operate on the air-liquid interface at laminar-to-laminar flow. Friccohesity scans consolidate the contribution of surface energy from each laminar layer, causing a frictional force on maximum adherence with the least and higher  $\eta$ . Friccohesity decreases on increasing 4f <sup>#e</sup> from  $La_2S_3$  to  $Ho_2S_3$  at 283.15 and 315.15 K unlike that at 298.15 K that expeditiously exchanges via FRET, establishing a continuity to mutually exchange the alignment of H2O between adhesive and cohesive forces (Fig. 20a). LGT slightly tunes LSNR activities except for  $Ho_2S_3$  with higher and the least SE. The  $\psi_{PE}$ compensates for the electronic cloud of  $\psi_{\text{UPE}}$  with the vibrant activities of FE (Fig. 20b). Friccohesity indicates LSNR and LGT systematisation on BSA capping unlike the and values with cohad on mutually equilibrating the binding and dispersing forces, respectively. The LSNRs and LGT both indicate CMT with BSA where the friccohesity mutual resonating energies transfer to normalise the cohad, the ratio of the cohesive and adhesive forces (Fig. 20c and d).

# 4.5. Friccohesity velocity ( $\sigma_v$ , cm s<sup>-1</sup>)

The disruption of LSNRs, GO, and LGT alone and with BSA causes reorientation with a certain velocity that could be decisive with respect to their adhesive and cohesive forces to mutual stabilization. The solvent molecules adhere around LSNRs and LGT surfaces, forming an NHS and to suspend them needs a strengthening of the structural framework with the required adhesion supported by mutual cohesion. The spontaneity of a template with respective  $\psi_s$  establishes a close continuity as an ordered state. Locally aligning and adhering a solvent induce collisions for continuous flow oscillation to maintain

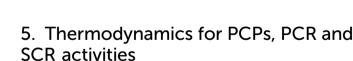
The  $\sigma_v$  values of LSNRs and LGT separately and with BSA at constant pH and temperature indicate alignment in capillary flow (Tables S5.9 and S6.0). The reorienting chemical species could initiate a solvent-disrupting activity to dislocate a free hydrogen-bonded state and adhere to LSNRs, GO, and LGT with specific velocity. Adhered H<sub>2</sub>O spontaneously align on the outer surface of the NHS, initiating the development of solvent–solvent coherence. The  $\sigma_v$  depicted as adherence and coherence move dynamically.

### 4.6. Isentropic compressibility $(K_{S,\phi})$

LSNRs, GO, LGT, and BSA reorient to adhere  $\rm H_2O$  from the hydrogen-bonded state with a higher cohesive force, causing monodispersion. Thereby charge expression plays a key role in generating entropy on reorienting and aligning them.  $K_{\rm S, \varphi}$  indicates constant entropy during these activities as the species maintain their structures, except for infinitesimal changes in friccohesity, for initiating interactions producing  $K_{\rm S, \varphi}$  in the  $5 \times 10^{-7}$  cm s<sup>2</sup> g<sup>-1</sup> range (Tables S7.5 and S7.6, ESI†).  $K_{\rm S, \varphi}$  is calculated using density and sound velocity (Tables S1.3, S1.8 and S8.2, ESI†):

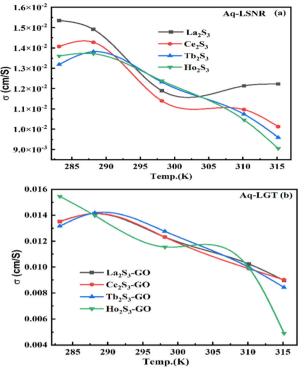
$$K_{s,\phi} = \frac{1}{u^2 \rho} \tag{8}$$

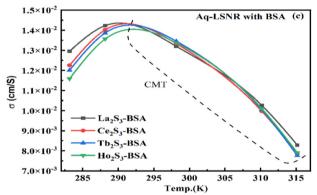
where u and  $\rho$  are sound velocity and density of solution. The  $K_{S,\varphi}$  of LSNRs and LGT alone and with BSA depict their electrostatic interaction with almost constant entropy due to the ordered alignment maintained by adherence and coherence within infinitesimal changes (Tables S7.5 and S7.6, ESI†). The  $K_{S,\varphi}$  of LSNRs and LGT gradually decrease with increasing temperature as it disrupts the ordered state. At higher temperature the  $\rho$  values are lower, as cohesive forces are weakened by oscillation of their binding energy. At lower temperature, a mild oscillation indicates a maximum  $K_{S,\varphi}$  but increasing temperature has weakened the  $\varepsilon$  of LSNRs and LGT with BSA (Fig. S1.5 and S1.6, ESI†). The increasing temperature indicates stronger LGT  $\sim$  BSA interaction with 4f  $^{\#e}$   $\sim$  FE  $\sim$  PB linkages due to the lowest friccohesity, as adherence is dominated by the cohesive forces of a uniform NHS.



# 5.1. $E_{\rm a}$ , kJ mol<sup>-1</sup>

LSNR and LGT dissolution need energy and space to disperse in  $\rm H_2O$ , so the HB disruption by weakening cohesive forces activates  $\rm H_2O$  alignment around them via electrostatic and ionic dipoles. LSNR and LGT dispersion acquire energy from water directly, indicating their interacting nature with a certain state function. The  $\rm Ln^{3+}$  and  $\rm S^{2-}$  of LSNRs vis-à-vis the skeletal edges of GO and their spatial occupation of the LGT interface encounter BSA with an active electronic cloud. These respond to





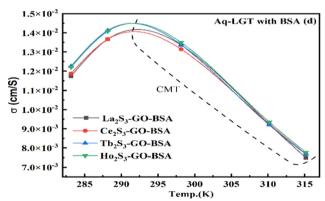


Fig. 20 Friccohesity of aq-LSNR and aq-LGT with BSA at 5 T.

cohesion, so  $\sigma_v$  is calculated with  $\sigma$  values (Tables 10 and 11).

Friccohesity velocity 
$$(\sigma_{\rm v}) = \frac{1}{\sigma \, {\rm s \, cm^{-1}}}$$
, or  $\sigma_{\rm v} \, {\rm cm \, s^{-1}}$  (7)

This article is licensed under a Creative Commons Attribution-NonCommercial 3.0 Unported Licence. Open Access Article. Published on 09 December 2022. Downloaded on 12/29/2025 6:35:29 PM.

interlinking of oscillations and thermal energies on increasing

$$\frac{p^2}{2m} = \frac{1}{2\pi} \sqrt{\frac{k}{\mu}} \tag{9}$$

The acquired energy does not break bonds, but rather activates a whole structure to develop  $\psi$  that resonates energy to participate in interactions.  $E_a$  is calculated with eqn (9) and  $\log \eta$  vs. (1/T) and log(abs) vs.  $(1/C_t)$  are plotted (Tables S1.4–S2.1, ESI†), where T is temperature and  $C_t$  is dye degradation concentration, with time tmin fitted as shown below:

$$\log(\eta) = \log A - \frac{E_{\rm a}}{R} \times \left(\frac{1}{T}\right), \text{ or slope} = -\frac{E_{\rm a}}{2.303 \times R},$$

$$E_{\rm a} = -\operatorname{slope} \times 2.303 \times R \tag{10}$$

 $E_a$  is calculated with dye degradation concentration  $(C_t)$  on PCR and SCR.

log(abs) = log 
$$A - \frac{E_a}{R} \times \left(\frac{1}{C_t}\right)$$
, or slope =  $-\frac{E_a}{2.303 \times R}$ , (11)  
 $E_a = -\text{slope} \times 2.303 \times R$ 

 $E_a > 0$  for Ln<sub>2</sub>S<sub>3</sub> and Ln<sub>2</sub>S<sub>3</sub>:GO indicates HB disruption realigning and reorienting H<sub>2</sub>O to flow in a streamlined way within a uniform capillary.  $(E_{a,LSNR} > E_{a,LGT}) > 0$  indicates higher activities of the FE despite it getting doped with LSNRs. Thus, the 4f #e of LGT resonates energy to transfer to the closest FE to photocatalyze dyes in a shorter time and is proven to be an efficient photocatalyst to induce FRET. Ea calculated with PCR is 10 times higher than with PCP, as multiple and manifold collisions of  $4f^{\#e} \sim FE$  utilize their energies to align (Fig. 21 and Tables S1.4-S2.1, ESI†) compared to shear stress being used to align.

### 5.2. Thermodynamic parameters $\Delta H$ , $\Delta G$ , and $\Delta S$

Thermodynamic functions define the interacting activities of LSNRs, GO, LGT, and BSA in aqueous solution as state functions  $\Delta H$ ,  $\Delta G$ , and  $\Delta S$  calculated from  $E_a$  (Tables S1.4–S2.1, ESI†):

$$\Delta H = E_a - 2.303RT \text{ or } \Delta H = E_a - 2.303RC_t$$
 (12)

$$\Delta G = 2.303RT \log(abs)$$
 or  $\Delta G = 2.303RC_t \log(abs)$ 
(13)

$$\Delta S = \frac{(\Delta H - \Delta G)}{T} \text{ or } \Delta S = \frac{(\Delta H - \Delta G)}{C_t}$$
 (14)

 $\Delta G$  < 0,  $\Delta H$  < 0 and  $\Delta S$  > 0 with LSNRs indicate spontaneously exothermic disordered interaction activities, reducing dyes in a longer time (Tables S1.4–S2.1, ESI†).  $\Delta H < 0, \Delta G > 0$ and  $\Delta S$  < 0 indicate an aligned LGT with H<sub>2</sub>O dipoles where non-spontaneous endothermic activity has reduced the dyes in shorter time, acquiring a lower  $E_a$  (Tables S8.3–S8.6, ESI†). The aligned and ordered LGT with  $\Delta G > 0$  has a stronger dipolar system ( $\Delta S < 0$ ), as the  $h\nu$  absorption ability is rapid for the PCR and SCR of dyes (Tables S4.2-S5.9, ESI†). Sonication disrupts the NHS and the secondary bonds release energy,

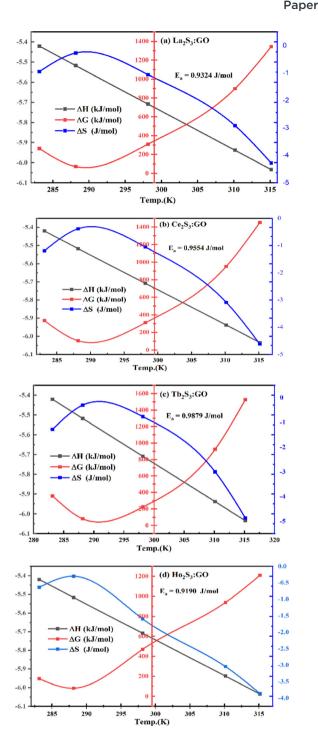


Fig. 21 The thermodynamic parameters with increasing temperature of Aq-LGT for PCR.

increasing the overall heat content as an exothermic process.  $E_a$ for LGT is the lowest, as the FE and 4f #e are both pre-activated with a least QEB of 4f<sup>#e</sup> and ordered FE, as  $\langle \psi_{4f^{\#e}} | \psi_{FE}^2 \rangle$  is a normalized quantum string on the surface. With a mutual contribution, the RE of FGs and 4f #e are partially shed to reorient and align GO around LSNRs is favoured by NHS mutually (Fig. 5b). LSNRs and LGT could not affect Ea with BSA, despite having many amino acids, tryptophan, salt bridges, and electrostatic linkages with H2O dipoles due to their structural integrity. Tb<sub>2</sub>S<sub>3</sub> with 4f<sup>9</sup> strongly aligns the H<sub>2</sub>O around BSA.

# 6. Photocatalytic reduction (PCR) of MB, MO, BBR and BBG under sunlight

The  $h\nu$  could interact with the electronic cloud available on the surfaces of LSNRs and LGT as an excessive cohesive force with maximum electronic sites to receive photons. Additionally, the packing of LSNRs, LGT, and GO photocatalyst causes a higher increasing rate of PCR. The monodispersion of photocatalyst generates a lower surface area with a maximum number of  $h\nu$ receiving sites. Therefore, the study of PCPs is noted as an essential prerequisite for monodispersion and to have thermodynamic and kinetic stability for the efficient use of active site of LSNR, GO, and LGT photocatalysts. Thus, the PCPs have been considered as preparatory ground for PCR and SCR. Fig. 22 depicts the homogeneous to heterogeneous phase; initially it dispersed in aqueous solution and after 90 min it was settled at the bottom, as these did not engage the dipoles of water without monodispersion. Thus, an adequate adherence of H<sub>2</sub>O is necessary for monodispersion for reusability, as the photon could not weaken the cohesive force for adhering H<sub>2</sub>O molecules, denoted as the secondary cohesive force. The primary cohesive forces do not monodisperse the sample and the secondary cohesive forces generate functional sites to enhance increasing rate of PCR and SCR activities. LSNRs and LGT both oscillate the electronic cloud to gain an electronic charge of dyes having quaternary nitrogen (QN) with electron-deficient sites through FRET. LSNRs and LGT both resonate their respective electronic charges to overcome the QEB to approach cycles. Their resonating energies counterbalance the respective residual charge without balancing the holes. GO holes out of hexagonal GO sheets PCR the MB, MO, BBR, and BBG with LGT. GO with a high mobility of charge carriers gains higher thermal conductivity to capture a maximum number of  $h\nu$ . <sup>20,35</sup> LGT has reduced dyes in  $\sim 30-90$  min and GO in  $\sim 120$  min with  $\Phi$  of 99–100 and 87%, respectively.<sup>3</sup> The @PCR and @SCR for dyes are calculated as:

@PCR/SCR% = 
$$\frac{C_0 - C_t}{C_0} \times 100$$
 (15)

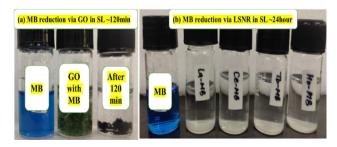
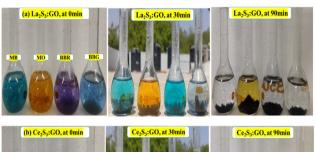


Fig. 22 PCR of MB: (a) aq-GO (b) aq-LSNR under SL in 120 min and 24 h, respectively







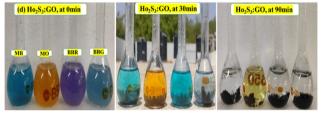


Fig. 23 PCR of dyes under SL with LGT ~30-90 min.

 $C_0$  is the initial dye concentration at time t = 0, and  $C_t$  is the concentration of reduced dye at t min. UV/Vis absorbance has been used to calculate the quantum yield using Einstein egn (16):

$$E = mc^2 (16)$$

$$abs = \varepsilon lc, \Delta c = \frac{\Delta Abs}{\varepsilon l}$$
 (17)

$$n_{\rm a} = \frac{E_{\rm a}}{h}, \ n_{\rm e} = \frac{E_{\rm e}}{h}, \ \Phi = \frac{n_{\rm e}}{n_{\rm e}}$$
 (18)

where m is the mass derived from eqn (17) with  $\Delta c$  the change in concentration,  $\Delta Abs$  is the change in absorbance, c is the speed of a photon  $(3 \times 10^8 \text{ m s}^{-1})$  and  $E_a$  is the absorbed energy and  $E_{\rm e}$  the emitted energy,  $n_{\rm a}$  is the number of photons absorbed and  $n_{\rm e}$  is the number of photons emitted. The  $h\nu$ from sunlight generated phonons different frequencies, so a particular energy is used to calculate the number of  $h\nu$ . Absorption is used to calculate  $C_t$ , which directly depends on number of  $h\nu$ , which is then used to calculate  $n_a$  and  $n_e$  (Fig. 24).

LGT has reduced BBG, BBR, MB and MO with 90-100, 87-97, 90-96, and 93-95% @PCR and with 94-100, 94-99, 93-97, and 93–95%  $\Phi$ , respectively due to their different structures (Tables S8.3.0-S8.6, ESI†) Tb<sub>2</sub>S<sub>3</sub>:GO and Ho<sub>2</sub>S<sub>3</sub>:GO are effective **Energy Advances** Paper

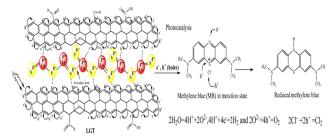


Fig. 24 LGT electronic hole generation for the PCR of dyes

photocatalysts with 4f 9e and 4f 11e, respectively. Photocatalysts and dyes both align mutually to exchange their electronic charges as  $\psi$  overcomes the QEB. MB with 4-CH<sub>3</sub> causes and ER effect on absorbing  $h\nu$  (Fig. 25c). The active sites of dyes interact with LSNRs and LGT, but QN receives electron holes from LSNRs and LGT through FRET (Fig. 25a). LSNRs and LGT distinguished the azo group of MO despite resistance to the conjugation of electrons (Fig. 25d). MO dye is reduced in a longer time than BBG, BBR, or MB as the hydrophobicity of -CH<sub>3</sub> influences the delocalisation of the  $\pi$  electrons. The binding constant (k<sub>b</sub>) of MO with LSNRs and LGT is higher with a lower reduction, indicating oscillation of MO to interact with the photocatalysts. The higher  $k_b$  blocks the  $h\nu$  receiving sites of the photocatalysts, inducing defects on interacting with them as well as with dyes. The -N=N- azo and Na<sup>+</sup>-SO<sub>3</sub><sup>-</sup> interact, reducing the surface area to become settled at the bottom; therefore, the PCPs and PCR are directly interfaced. ERG (-CH<sub>3</sub>) intensifies their holes for reducing BBG and BBR in a shorter time than MO with LGT and LSNRs, initiating an ROC  $via h\nu$ . The QNs of BBG and BBR received the e<sup>-</sup> holes from LGT  $via \ \psi$  for the PCR (Fig. 25c and d). BBR, BBG, MB, and MO are reduced at  $\lambda_{\text{max}}$  of 550, 585, 665, and 465 nm, respectively, with UV/vis, light and sunlight for disposal. The reduced dyes were phased out from wastewater for recycling (Fig. 23). The  $h\nu$ weakened the shared electron and FGs accompanied by certain thermodynamic parameters.  $\Delta H < 0$ ,  $\Delta G > 0$  and  $\Delta S < 0$  as an exothermically spontaneous reaction has fixed the @PCR and (a) SCR of dyes. The  $h\nu$  have struck FE and 4f <sup>#e</sup> on decreasing S compared to aq-LSNR to minimize the collision of holes.

The absorbance of wastewater effluents decreases with the 4f electrons of the photocatalyzing properties of the photocatalyst. An 87-100% reduction of dyes was observed after  $\sim$  60 min. The enhanced photocatalyzing performance of LGT was noticed due to a higher amount of UV light being detained and higher photon trapping ability. The catalysing activities were observed by plotting  $(C_{4\text{fe}}/C_0)$  vs.  $4f^{\text{#e}}$ , to observe the role of electronic activities during PCR (Fig. 26). The @4f#e of the photocatalyst indicates the excellent photocatalyzing ability extended to determine the role of the maximum number of 4f electron in the lanthanide series.

# 6.1. Reusability of LGT photocatalyst

The reusability of a photocatalyst is studied by using the same photocatalyst over 4 to 5 cycles to reduce the fresh effluents. Every cycle reduced the active sites of the photocatalyst, as the

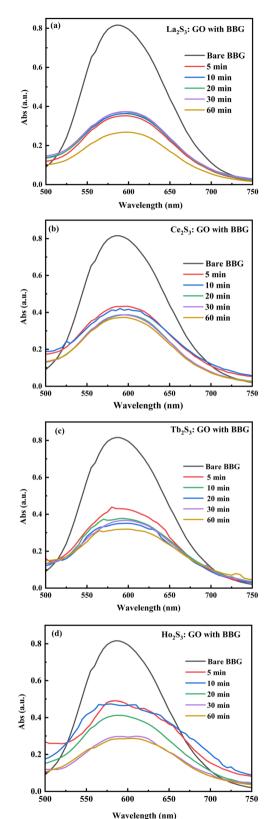


Fig. 25 Photocatalysis of (a) BBR, (b) BBG, (c) MB and (d) MO through LGT.

quantum yield of the dyes decreases with successive cycles. PCR performance has not significantly decreased after 4-5 cycles

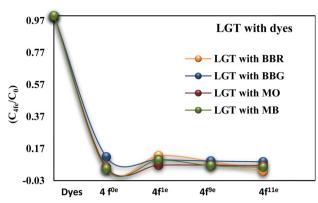


Fig. 26 Photocatalytic activity as a function of 4f electrons

due to the structural stability inducing the reusability of the photocatalyst, as supported by TGA on the disruption of the lattice (Fig. 1). Reusability of photocatalysts implies a decrease in active sites in subsequent cycles and mass loss of photocatalyst over time (Fig. 28). The wastewater effluents were suspended in solution at the 4th cycle and settled at the bottom after 1 h as the active sites of the photocatalyst have been disrupted and do not respond to dyes any more (Fig. 28).

Due to ERG-CH<sub>3</sub> BBG has responded exceptionally to overcome a QEB of holes of the GO and 4f 8e of Tb3+. The conduction band has 4f 6e unpaired electron (UNPE) which are robustly excited to reach a conduction band of 5d and then an electron partially comes back and a phonon is emitted. That approaches the excited electron of a dye, especially its reducing site, and it gets reduced. The LGT of the LSNRs followed the same trends,

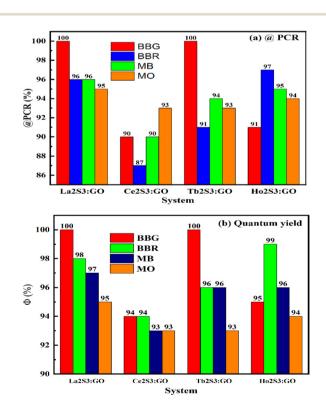
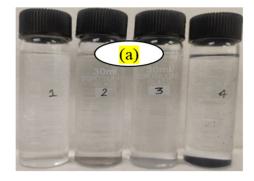


Fig. 27 (a) @PCR and (b) quantum yield for PCR under SL.



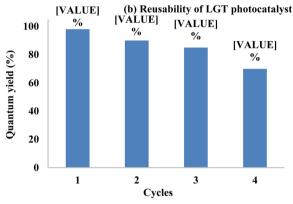


Fig. 28 (a) Recycled wastewater. (b) Reusability of photocatalyst LGT.

which have proven the 4f 6e UNPE of Tb to robustly interact with FE to catalyse the generation of holes. These holes reach the dyes in a larger amount in a shorter time, because 4f 6e could have gained similar energy from the  $h\nu$  to get them excited with a similar  $\varphi$  to PCR the dyes in a shorter time than other LGT. Ho<sup>3+</sup> has 4f <sup>4e</sup> UNPE which could not induce an impact on the  $h\nu$  vis-àvis dye to generate holes, and so took a bit longer to reduce the same dye compared to the LGT of Tb with 4f <sup>6e</sup> UNPE. A similar action mechanism is followed for Ce2S3:GO and La2S3:GO as for the UNPE generating phonons. The @4f #e UNPE directly interact with the electron which could get excited to interact with the holes. Hence, the higher DOS of increasing 4f #e UNPE interacts with the intensified DOS of the dyes to get reduced as BBR > BBG > MB > MO and with increasing time as MO > MB > BBG > BBR. Thereby, the BBR and BBG with adequate ERG-CH<sub>3</sub> as well as maximum  $\pi$ -conjugation have taken a shorter time with each LSNR and LGT (Fig. 28). This process is denoted as @4f<sup>#e</sup>, but only the DOS of the dye develops an interacting interfacial energy to overcome the QEB in a shorter time. The MO dye has azo and SO3- groups, which generate their own intensified and localized  $\psi(SO_3^-)$  and  $\psi(N=N)$  that could not synchronize with the  $\pi$ -conjugation of the two benzene rings of MO dye. The EEC and EER have both neutralized the energy of the dyes on receiving the  $h\nu$  that has taken a longer time, despite the presence of LSNRs and LGT. Therefore, the electronic constituents of the dyes matter a lot for reduction as the MO have DOS of  $\psi(SO_3^-)$ and  $\psi(N=N)$  unlike the DOS of the  $\psi_{Benzene}$  ring expressed as  $\psi(SO_3^-) \neq \psi(N=N) \neq \psi_{Benzene}$ . Thus, the DOS of BBR and BBG could have been similar except for the ERG with BBG which take a

shorter time. Additionally, the MB due to two different ethyl groups with an almost nil effect of ERGs could not develop an intensified DOS and has taken more time for reduction than BBR or BBG (Fig. 22).

# 7. Sonocatalytic reduction (SCR) of MB, MO, BBR and BBG

LSNRs, GO, and LGT were not homogeneously dissolved in water and phased out without electrostatically engaging the dipoles of water to transfer a cohesive force as PE to oscillatory energy. The  $h\nu$  was not captured by photocatalysts which were sonicated for monodispersion to SCR the dyes by generating functional sites. The sonication generates localized energy centres denoted as charge cavitation. The charge cavitation gains a higher surface area and available electronic oscillatory or resonating activity to develop  $\psi$  that cross over a QEB of electronic charge sites of dyes, and the species remained monodispersed. The sonication was further extended from monodispersion to generate photonic waves with a certain  $h\nu$ , equally capable of reducing dyes. Therefore, in both PCR and SCR, a monodispersion of photocatalyst and dyes is a prerequisite. Therefore, the PCPs as a database predict thermodynamic and kinetically stable states for receiving hv to generate ROC. SCR was conducted to equate PCR efficiency with monodispersion and then the  $h\nu$  was supplied and sonication continued to SCR. Sonication weakened the PE of shared electrons and FGs with both oscillating as charged clouds on a cavitation that aligns the electronic cloud shifting to a higher energy state to generate e and h holes. The sonication induces mechanical energy to generate an electron-rich domain e or electronic-charge-deficient h holes. LGT, LSNRs and GO solutions with dyes dissociated their bonds to move the electrons towards the respective holes of dyes for SCR. 48 Sonication could also break the bonds and release energy due to oscillations.<sup>39</sup>  $\Delta S$  < 0 and  $\Delta S$  > 0 indicate ordered and disordered configurations, and oscillations of LGT might be mildly disordered on aligning dyes with 40-90%  $\Phi$ (Tables S7.7-S8.0, ESI†). The SCR route is effective for dipolar LSNRs and LGT by sonicating at 28 KHz for 5, 10, 20, 30, and 60 min, consequently raising the temperature to 298.15, 301.15, 305.15, 308.15, and 313.15 K, respectively. SCR structurally

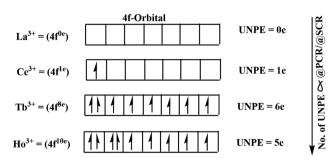


Fig. 29 The number of unpaired electrons directly enhances the @PCR and @SCR.

reduces a dye as a sono-frequency attacks the  $\pi$ -conjugation to enhance the electronic charge density. The structurally damaged dye does not respond to environmental and biological processes. A highly oxidised structure with  $\Delta S < 0$  is phased out (Tables S4.2-S5.8, ESI†).  $\Delta G > 0$  indicates interaction between dyes and LGT on sonication. Ea is not needed for

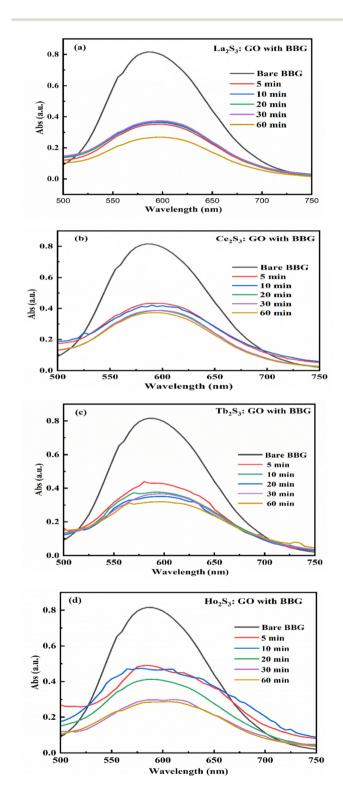


Fig. 30 SCR of BBG with LGT and other dyes given (Fig. S8-S11, ESI†).

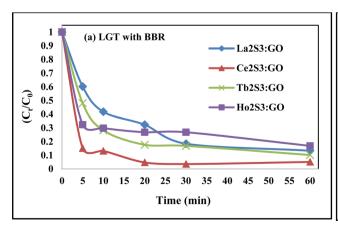
SCR, as adequate energy already exists.  $\Delta G > 0$  values indicate that the sonicating frequency drastically disrupts a shared electron pair of covalent bonds (Fig. S2.4–S2.7, ESI†). The  $\Delta G$ indicates simultaneous oscillation by a molecule to gain  $\Delta S < 0$ with  $\Delta H < 0$  values.  $\Delta H < 0$  and  $\Delta S < 0$  while the  $\Delta G > 0$  for all the LSNR systems with BBR, BBG and MB, except MO, have values  $\Delta S > 0$  (Fig. 29, and Fig. S2.4, S3.1, S3.2, ESI†).  $\Delta S > 0$  of MO is strongly disordered due to an azo group with the least @SCR. The excess π-conjugation increases the ability of @SCRand @PCR due to collisions and ROC (Fig. 23b).  $\Delta S \ll 0$  with the least disorder has a higher @SCR. The least disordered and maximum @SCR as BBR > BBG > MB > MO indicate alignment (Tables S4.2-S5.8, ESI†). SCR in the order BBR > MB > BBG > MO with binding constants  $(k_b)$  0.0211 < 0.0255 < 0.0658 < 0.0800, respectively, indicate restricted MO activities as the binding causes a PCR and SCR. The oxidized  $\approx$  reduced on binding dye with LSNRs or LGT as

$$k_{\rm b} = \frac{\rm LGT/LSNR - Dye_{r/o}}{\rm LGT/LSNR - Dye_{o/r}}$$
 (19)

Eqn (19) indicates LGT/LSNR–Dye<sub>o/r</sub>  $\neq$  LGT/LSNR–Dye<sub>r/o</sub> with an equilibrium state. The higher  $k_{\rm b}$  for MO indicates a lower @PCR and @SCR as an azo group causes many collisions. The @PCR and @SCR for BBR, BBG and MB are maxima with a lower  $k_{\rm b}$  as a monodispersion. BBR, BBG and MB with a maximum  $\pi$ -conjugation is sensitive, unlike MO with weaker @SCR activity (Fig. 30).

The dye alone absorbed maximum UV light at  $\lambda_{\rm max}$  at 550, 585, 665, and 465 nm for BBR, BBG, MB, and MO, respectively, in both PCR and SCR, due to the structural stability of LSNRs and LGT predicted by the PCPs. The reduction of dyes at the same  $\lambda_{\rm max}$  indicates no degradation of dyes into fragments and also no specific complex formation between the dyes and photocatalysts. However, the broadening of the  $\lambda_{\rm max}$  peaks indicates the distribution of similar species forming or yet to be reduced. Thus, the PCPs predict a stronger monodispersion of LSNRs, GO, and LGT. Absorbance of dyes decreases with time and 4f <sup>#e</sup> during an SCR. The time and 4f <sup>#e</sup> as vital parameters analyse the ability of a photocatalyst to reduce a

dye. The active sites of the dye were trying to capture the opposite holes with substantial sonication energy and get reduced. The sonocatalyzing performance is observed by plotting  $(C_{4\text{fe}}/C_0)$  vs.  $4f^{\text{#e}}$  (Fig. 31). The increasing  $4f^{\text{#e}}$  induced SCR activities to catalyse wastewater effluents. The time-dependent thermodynamic parameters for LSNRs, LGT with BBR, BBG and MO under SCR indicate reduction activities (Fig. S3.1 and S3.2, ESI†). The increasing rate of sonication energy has electronically disordered a structure with  $\psi$  of LGT oscillation. Compared to PCR, the SCR streamlined the dipoles and reduced dyes with 40–90%  $\Phi$ . The  $\Delta S$  with dyes from La<sub>2</sub>S<sub>3</sub> to La<sub>2</sub>S<sub>3</sub>:GO decreases as La<sub>2</sub>S<sub>3</sub>:GO is highly ordered and partially oscillates due to valence electrons (Fig. S3.1, ESI†). The  $\Delta S$  with La<sub>2</sub>S<sub>3</sub> to Ce<sub>2</sub>S<sub>3</sub>:GO increases as 4f 1e oscillates, but Tb<sub>2</sub>S<sub>3</sub>:GO to  $Ho_2S_3$ :GO are ordered with a least  $\Delta S$ , so BBR has interacted with 4f <sup>#e</sup> and FE via  $\pi$ -conjugation (Fig. S3.1, ESI†). Tb<sub>2</sub>S<sub>3</sub>:GO to  $\text{Ho}_2\text{S}_3$ :GO with  $\Delta S < 0$  due to stable dipoles SCR dyes in shorter time. LSNRs with higher  $\Delta S$  than LGT but could not SCR MO due to an azo group compared to LGT. LSNRs have disordered an alignment of dipoles causing recombination of holes. Sonication to induce undesired collisions with @SCR as the dipoles attract the holes of GO. BBG with -CH3 could not establish patterns of  $\Delta S$  values causing higher @SCR (Fig. S3.1, ESI†).  $\Delta S$  values are scattered with MO unlike BBG due to -CH<sub>3</sub> with different electronic junctions causing SCR (Tables S4.2-S5.8, ESI†).  $\Delta G$  values for MB, BBR, and MO with increasing time are increased except the BBG as it has an extra ER group that disrupts a bond on a sonication (Fig. S2.4-S2.7, ESI†). Insoluble LSNRs and LGT could not SCR itself with  $\Delta G > 0$ , but MB with LSNRs having  $\Delta G < 0$  indicates an LSNR  $\sim$  MB solubilizing combination with  $H_2O$  while LGT having  $\Delta G$  > indicates weaker spontaneity (Tables S4.2–S5.8, ESI†). MO with  $\Delta G < 0$ indicates a stronger  $k_b$  of azo dye, and a milder sonication effect occurring than for other dyes (Fig. S2.7d, ESI†). The  $\Delta G > 0$ and  $\Delta H < 0$  of LSNRs, GO, and LGT indicate a stronger binding with BBG due to -CH<sub>3</sub> (Tables S4.2-S5.8, ESI†). Sonication breaks the NHS and splits  $H_2O$ , releasing  $\Delta H$ , and fragments become vibrant (Fig. 27). BBG and BBR have released a higher  $\Delta H$  than MB and MO to align LSNRs, GO and LGT and causing the higher SCR activities (Tables S4.2-S5.8, ESI†).



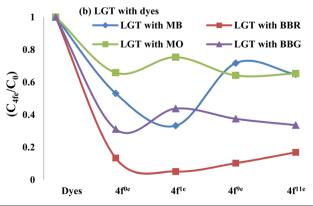


Fig. 31 (a) Sonocatalytic activity as a function of time and (b) as a function of 4f electrons.

# 8. Tables

 Table 5
 Dynamic light scattering analysis of GO, LSNRs and LGT aqueous solution

Parameter	GO	$La_2S_3$	La <sub>2</sub> S <sub>3</sub> :GO	$Ce_2S_3$	Ce <sub>2</sub> S <sub>3</sub> :GO	$Tb_2S_3$	Tb <sub>2</sub> S <sub>3</sub> :GO	$Ho_2S_3$	Ho <sub>2</sub> S <sub>3</sub> :GO
PDI	_	0.425	1.591	1.002	0.175	0.619	1.023	5.35	1.369
Diameter (nm)	6.10	457.0	307	17.66	263.7	486	212.9	16.57	325
,	_	230.3	_	_	109.3	_	97.60	_	157.7
Volume (%)	100	74.8	100	100	47.2	100	52.1	100	20.9
	_	25.2	_	_	52.8	_	47.9	_	79.1
Width	2.32	344.00	158.70	9.28	161.10	169.30	126.80	6.18	98.00
	_	54.90	_	_	44.40	_	36.50	_	104.30
Mobility ( $\mu \text{ s}^{-1}$ )	0.81	1.74	-0.04	-0.04	1.22	0.30	-1.40	0.62	0.87
Zeta pot. (mV)	10.41	22.24	-0.49	-0.51	15.64	3.88	-17.60	7.97	11.08
Charge (fC)	0.00052	0.327	-0.004	-0.0064	0.033	0.036	-0.057	0.059	0.032
Polarity	+ve	+ve	-ve	-ve	+ve	+ve	-ve	+ve	+ve
Conductivity (μs cm <sup>-1</sup> )	50	43	18	27	17	24	20	38	22

Table 6 Viscosities (mPa S) of LSNRs and LGT

Temp. (K)	$La_2S_3$	$Ce_2S_3$	$Tb_2S_3$	$Ho_2S_3$	La <sub>2</sub> S <sub>3</sub> :GO	Ce <sub>2</sub> S <sub>3</sub> :GO	Tb <sub>2</sub> S <sub>3</sub> :GO	Ho <sub>2</sub> S <sub>3</sub> :GO
283.15	1.0152	0.9035	0.8705	0.8514	0.8944	0.8680	0.8596	0.9281
288.15	1.0475	0.9736	0.9565	0.9497	0.9704	0.9565	0.9668	0.9668
298.15	0.8746	0.8787	0.8827	0.8945	0.8825	0.8827	0.9142	0.8275
310.15	0.7095	0.6986	0.6986	0.6950	0.7058	0.6913	0.6986	0.6950
315.15	0.6379	0.5930	0.5833	0.5650	0.5990	0.5748	0.5586	0.6301

Table 7 Viscosities (mPa S) of LSNRs and LGT with BSA

Temp. (K)	BSA	$La_2S_3$ BSA	Ce <sub>2</sub> S <sub>3</sub> BSA	$Tb_2S_3$ BSA	Ho <sub>2</sub> S <sub>3</sub> BSA	La <sub>2</sub> S <sub>3</sub> :GO BSA	Ce <sub>2</sub> S <sub>3</sub> GO BSA	Tb <sub>2</sub> S <sub>3</sub> :GO BSA	Ho <sub>2</sub> S <sub>3</sub> :GO BSA
283.15	0.8699	0.8668	0.8837	0.8727	0.8617	0.8638	0.8699	0.8973	0.8669
288.15	0.9634	0.9630	0.9702	0.9668	0.9531	0.9599	0.9600	0.9770	0.9599
298.15	0.8827	0.8824	0.8788	0.8866	0.8748	0.8866	0.8788	0.8747	0.8826
310.15	0.6367	0.6364	0.6403	0.6402	0.6439	0.6465	0.6474	0.6437	0.6543
315.15	0.4945	0.4937	0.5067	0.4977	0.5099	0.5054	0.5128	0.5157	0.5209

Table 8 Surface tension (mN m<sup>-1</sup>) of LSNRs and LGT

Temp. (K)	$La_2S_3$	$Ce_2S_3$	$\mathrm{Tb}_2\mathrm{S}_3$	$Ho_2S_3$	La <sub>2</sub> S <sub>3</sub> :GO	Ce <sub>2</sub> S <sub>3</sub> :GO	Tb <sub>2</sub> S <sub>3</sub> :GO	Ho <sub>2</sub> S <sub>3</sub> :GO
283.15	65.97	63.82	66.01	65.56	66.01	64.25	65.11	67.41
288.15	70.25	68.19	69.21	69.21	68.72	67.70	68.19	69.21
298.15	71.06	71.64	71.64	72.22	71.63	71.64	71.64	71.64
310.15	58.481	63.725	65.07	66.48	68.95	67.44	68.95	70.00
315.15	51.98	58.38	60.56	62.15	66.52	63.47	66.01	67.99

Table 9 Surface tension (mN m<sup>-1</sup>) of LSNRs and LGT with BSA

Temp. (K)	BSA	La <sub>2</sub> S <sub>3</sub> BSA	Ce <sub>2</sub> S <sub>3</sub> BSA	Tb <sub>2</sub> S <sub>3</sub> BSA	Ho <sub>2</sub> S <sub>3</sub> BSA	La <sub>2</sub> S <sub>3</sub> :GO BSA	Ce <sub>2</sub> S <sub>3</sub> :GO BSA	Tb <sub>2</sub> S <sub>3</sub> :GO BSA	Ho <sub>2</sub> S <sub>3</sub> :GO BSA
283.15	74.21	66.91	71.97	72.52	74.20	73.58	73.07	73.06	70.88
288.15	70.799	67.67	69.21	69.73	70.26	70.26	70.26	69.21	68.19
298.15	66.868	66.84	65.89	65.89	65.41	66.37	66.87	65.41	65.40
310.15	65.995	61.98	64.16	63.71	63.71	65.89	65.97	65.50	65.47
315.15	67.00	59.36	64.27	63.86	64.27	67.28	66.51	66.95	66.83

**Table 10** Friccohesites (cm  $S^{-1}$ ) of LSNRs and LGT

Temp. (K)	$La_2S_3\left(10^{-4}\right)$	$Ce_2S_3$ (10 <sup>-4</sup> )	$Tb_2S_3 (10^{-4})$	$Ho_2S_3 (10^{-4})$	$\text{La}_2\text{S}_3$ :GO (10 <sup>-4</sup> )	$Ce_2S_3$ : $GO(10^{-4})$	$Tb_2S_3$ : $GO(10^{-4})$	${ m Ho_2S_3:GO~(10^{-4})}$
283.15	154	140	132	136	135	135	132	155
288.15	149	143	138	137	141	141	142	140
298.15	119	114	123	124	123	123	128	116
310.15	121	110	107	105	102	099	100	099
315.15	122	101	096	091	090	090	084	049

Table 11 Friccohesity (cm S<sup>-1</sup>) of LSNRs and LGT LGT with BSA

	_
283.15 117 130 122 120 116 117 119 123 122	
288.15 136 142 140 139 136 137 137 141 141	
298.15 132 132 133 135 134 134 131 134 135	
310.15 096 103 100 100 101 092 092 092 093	
315.15 074 083 079 078 079 075 077 077 078	

MB

**PCR** 

**ERG** 

TGA

DTG

Methylene blue

# 9. Conclusion

Multifunctional LSNR and LGT photocatalysts and adsorbents were prepared and had enhanced solubility of BSA by initiating interactions with the peptide bonds and electrostatic dipoles of aq-BSA with solubilizing activities. Physicochemical properties and thermodynamic functions of aq-LSNR, and aq-LGT alone and with 0.010 g% BSA at 283.15-315.15 K and within 0.002-0.01 g% @0.004 g% at 298.15 K indicating structural stability by doing PCR and SCR activities. The physicochemical properties and thermodynamic functions of BSA with LSNRs and LGT indicates a stronger  $4f^{\#e} \sim FE$  interface with electrostatic dipoles, peptide bonds, and salt bridges as a robust approach in the area of lanthanide chemistry with biomolecules. The stronger temperature and concentration effect with structural stability, functionality, and thermal stability of LSNRs and LGT enabling their reusability for PCR and SCR activities. LSNRs with a stronger  $h\nu$  trapping ability to photocatalyze the fluorescent dyes in a longer time while LGT has shortened reduction time due to 4f#e ~ FE interfaces. LGT recycled wastewater containing fluorescent effluents in ~60 min compared to LSNRs and GO with  $\sim 24$  h and 120 min, respectively. LGT has reduced BBG, BBR, MB, and MO as 90-100, 87-97, 90-96, and 93-95% @PCR with 94-100, 94-99, 93-97, and 93-95% quantum yields, respectively. Each dye was sonicated @28 kHz with LSNRs and LGT separately for 5, 10, 20, 30, and 60 min were consequently raised to 298.15, 301.15, 305.15, 308.15, and 313.15 K, respectively, as an exothermic activity of LSNRs and an LGT support with thermodynamic functions. SCR has reduced the dyes with 40 to 90%  $\Phi$ . PCR and SCR activities indicated that LGT is a good photocatalyst and a solubilizing agent with excellent reusability.

# **Abbreviations**

GO Graphene oxide

LSNRs Lanthanide sulphide nanorods LGT Lanthanide GO template

BBR	Brilliant blue red
BBG	Brilliant blue green
MO	Methylene orange
BSA	Bovine serum albumin
PCPs	Physicochemical properties
FT-IR	Fourier transform infrared spectroscopy
GT	Graphite
GtO	Graphite oxide
RB	Round bottom flask
RPM	Revolutions per minute
aq	Aqueous
PDN	Pendant drop number
VFT	Viscous flow time
wt	Weight
SCR	Sonocatalytic reduction
FE	Functional edge
RE	Resonating energy
HHI	Hydrophilic-hydrophilic interactions
HB	Hydrogen bond
PB	Peptide bond
CAT	Critical aggregation temperature
SL	Sunlight

# Author contributions

Photocatalytic reduction

Electron releasing group

Differential thermal gravimetry

Thermogravimetric

Krishan Kumar conducted the bench work, calculations, plotting figures and other designs for effective discussion and conclusive. Bhargab Sahoo initiated formulating the molecular solution preparation for accurate stoichiometry to measure the primary data of binary systems. Tara Chand Meghwal initiated formulating the molecular solution preparation for accurate stoichiometry to measure the primary data of ternary systems. Prof. Man Singh was in charge of overall supervision for

conducting the research work, frequent discussions, exploring new models outer experiment observations and matching with the explored scientific exhalations to substantially create the new science out of the research. Funding Sources There was no funding source.

# Conflicts of interest

The authors confirm that there is no conflict of interest.

# Acknowledgements

The authors thankful to Central University of Gujarat, India for facilities. Mr Krishan Kumar is thankful to UGC-DAE CSR Indore, was acknowledged for financial support (ref. CRS/ 2021-22/379) and Dr Mukul Gupta as a Principal Collaborator, and Dr Vasant Sathe, UGC-DAE CSR Indore, was acknowledged for providing Raman facility. The authors are thankful to CSIR-CMCRI Bhavnagar, Gujarat for FESEM + EDS analysis.

# References

- 1 X. Zeng, Y. Liu, T. Zhang, J. C. Jin, J. L. Li, Q. Sun, Y. J. Ai, M. L. Feng and X. Y. Huang, Ultrafast and selective uptake of Eu3+ from aqueous solutions by two layered sulfides, Chem. Eng. J., 2021, 420, 127613.
- 2 R. Kumar, S. Sahoo, E. Joanni, R. K. Singh, K. Maegawa, W. K. Tan, G. Kawamura, K. K. Kar and A. Matsuda, Heteroatom doped graphene engineering for energy storage and conversion, *Mater. Today*, 2020, **39**, 47–65.
- 3 S. Dev and M. Singh, Metallic sulfide nanoparticles anchored graphene oxide: Synthesis, characterization and reduction of methylene blue to leuco methylene blue in aqueous mixtures, J. Phys. Chem. Solids, 2020, 139, 109335.
- 4 A. T. Smith, A. M. LaChance, S. Zeng, B. Liu and L. Sun, Synthesis, properties, and applications of graphene oxide/ reduced graphene oxide and their nanocomposites, Nano Mater. Sci., 2019, 1, 31-47.
- 5 K. Kumar, R. P. Dave, S. Dev and M. Singh, Study of molar properties of GO after doping with transition metals for photodegradation of fluorescent dyes, RSC Adv., 2022, 29734-29756.
- 6 J. Dadashi, M. A. Ghasemzadeh and M. Salavati-Niasari, Recent developments in hydrogels containing copper and palladium for the catalytic reduction/degradation of organic pollutants, RSC Adv., 2022, 12, 23481-23502.
- 7 S. Roméro, A. Mosset, J. C. Trombe and P. Macaudière, Lowtemperature process of the cubic lanthanide sesquisulfides: Remarkable stabilization of the  $\gamma$ -Ce2S3 phase, J. Mater. Chem., 1997, 7, 1541-1547.
- 8 İ. A. Kariper, Synthesis and characterization of cerium sulfide thin film, Prog. Nat. Sci.: Mater. Int., 2014, 24(6), 663-670.
- 9 İ. A. Kariper, The production of UV absorber amorphous cerium sulfide thin film, Mater. Res., 2017, 20, 1345-1349.

- 10 C. Liu, Z. Yu, D. Neff, A. Zhamu and B. Z. Jang, Graphenebased supercapacitor with an ultrahigh energy density, Nano Lett., 2010, 10, 4863-4868.
- 11 G. Darabdhara, M. R. Das, S. P. Singh, A. K. Rengan, S. Szunerits and R. Boukherroub, Ag and Au nanoparticles/reduced graphene oxide composite materials: Synthesis and application in diagnostics and therapeutics, Adv. Colloid Interface Sci., 2019, 271, 101991.
- 12 M. Ubaidullah, A. M. Al-Enizi, T. Ahamad, S. F. Shaikh, M. A. Al-Abdrabalnabi, M. S. Samdani, D. Kumar, M. A. Alam and M. Khan, Fabrication of highly porous N-doped mesoporous carbon using waste polyethylene terephthalate bottle-based MOF-5 for high performance supercapacitor, J. Energy Storage, 2021, 33, 102125.
- 13 A. Roca-Sabio, M. Regueiro-Figueroa, D. Esteban-Gómez, A. de Blas, T. Rodríguez-Blas and C. Platas-Iglesias, Density functional dependence of molecular geometries in lanthanide(III) complexes relevant to bioanalytical and biomedical applications, Comput. Theor. Chem., 2012, 999, 93-104.
- 14 T. Mirkovic, E. E. Ostroumov, J. M. Anna, R. van Grondelle, Govindjee and G. D. Scholes, Light absorption and energy transfer in the antenna complexes of photosynthetic organisms, Chem. Rev., 2017, 117(2), 249-293.
- 15 X. Wei, H. Feng, L. Li, J. Gong, K. Jiang, S. Xue and P. K. Chu, Synthesis of tetragonal prismatic γ-In<sub>2</sub>Se<sub>3</sub> nanostructures with predominantly {110} facets and photocatalytic degradation of tetracycline, Appl. Catal., B, 2020, 260, 118218.
- 16 R. Tondwal and M. Singh, Chitosan functionalization with a series of sulfur-containing α-amino acids for the development of drug-binding abilities, J. Appl. Polym. Sci., 2018, 135, 1-14.
- 17 S. Mahapatra, G. Madras and T. N. G. Row, Ind. Eng. Chem. Res., 2007, 46, 1013-1017.
- 18 Z. Li, T. M. Rayder, L. Luo, J. A. Byers and C. K. Tsung, Aperture-Opening Encapsulation of a Transition Metal Catalyst in a Metal-Organic Framework for CO<sub>2</sub> Hydrogenation, J. Am. Chem. Soc., 2018, 140, 8082-8085.
- 19 P. Nuengmatcha, S. Chanthai, R. Mahachai and W. C. Oh, Sonocatalytic performance of ZnO/graphene/TiO2 nanocomposite for degradation of dye pollutants (methylene blue, texbrite BAC-L, texbrite BBU-L and texbrite NFW-L) under ultrasonic irradiation, Dyes Pigm., 2016, 134, 487-497.
- 20 M. Sun, J. Ma, M. Zhang, Y. Xiao, Y. Zhu and S. Zhang, Thiourea-modified Fe<sub>3</sub>O<sub>4</sub>/graphene oxide nanocomposite as an efficient adsorbent for recycling Coomassie brilliant blue from aqueous solutions, Mater. Chem. Phys., 2020, 241, 122450.
- 21 C. M. Marin, L. Wang, J. R. Brewer, W. N. Mei and C. L. Cheung, Crystalline α-Sm2S3 nanowires: Structure and optical properties of an unusual intrinsically degenerate semiconductor, J. Alloys Compd., 2013, 563, 293-299.
- 22 J. Choi, J. C. Kim, Y. B. Lee, I. S. Kim, Y. K. Park and N. H. Hur, Fabrication of silica-coated magnetic nanoparticles with highly photoluminescent lanthanide probes, Chem. Commun., 2007, 1644-1646.
- 23 D. G. Papageorgiou, I. A. Kinloch and R. J. Young, Mechanical properties of graphene and graphene-based nanocomposites, Prog. Mater. Sci., 2017, 90, 75-127.

24 O. P. Kumar, M. N. Ashiq, M. Ahmad, S. Anjum and A. Ur Rehman, Silver and yttrium-doped bismuth vanadate for photoluminescent activity and boosted visible light-induced photodegradation, J. Mater. Sci. Mater. Electron., 2020, 31, 21082-21096.

- 25 S. Aime, S. G. Crich, E. Gianolio, G. B. Giovenzana, L. Tei and E. Terreno, High sensitivity lanthanide(III) based probes for MR-medical imaging, Coord. Chem. Rev., 2006, 250, 1562-1579.
- 26 P. Macaudie, Low-temperature process of the cubic lanthanide sesquisulfides: remarkable stabilization of the c-Ce S phase, J. Mater. Chem., 1997, 7, 1541-1547.
- 27 V. J. Mane, S. B. Kale, S. B. Ubale, V. C. Lokhande, U. M. Patil and C. D. Lokhande, Lanthanum sulfide-manganese sulfide/graphene oxide (La<sub>2</sub>S<sub>3</sub>-MnS/GO) composite thin film as an electrocatalyst for oxygen evolution reactions, J. Solid State Electrochem., 2021, 25, 1775-1788.
- 28 J. Wu, K. Fujii, M. Yashima, A. Staykov, T. Akbay, T. Ishihara and J. A. Kilner, A systematic evaluation of the role of lanthanide elements in functional complex oxides; implications for energy conversion devices, J. Mater. Chem. A, 2018, 6, 11819-11829.
- 29 R. Chetty, S. R. Pandya and M. Singh, Physicochemical interaction of cerium oxide nanoparticles with simulated biofluids, hemoglobin, insulin, and ds-DNA at 310.15 K, New J. Chem., 2020, 44, 1825-1845.
- 30 J. Yin, Y. Hu and J. Yoon, Fluorescent probes and bioimaging: Alkali metals, alkaline earth metals and pH, Chem. Soc. Rev., 2015, 44, 4619-4644.
- 31 D. Kumar, A. Chandra and M. Singh, Influence of Urea on Shifting Hydrophilic to Hydrophobic Interactions of Pr(NO<sub>3</sub>)<sub>3</sub>, Sm(NO<sub>3</sub>)<sub>3</sub>, and Gd(NO<sub>3</sub>)<sub>3</sub> with BSA in Aqueous Citric Acid: A Volumetric, Viscometric, and Surface Tension Study, J. Chem. Eng. Data, 2014, 59(11), 3643-3651.
- 32 K. A. Gschneidner, Preparation and Processing of Rare Earth Chalcogenides, J. Mater. Eng. Perform., 2000, 7, 656-660.
- 33 Y. Han, Preparation of Calcium lanthanum Sulfide Powders via Alkoxide Su, J. Am. Ceram. Soc., 1991, 19, 2815-2819.
- 34 A. B. Ghosh, N. Saha, A. Sarkar, D. N. Srivastava, P. Paul and B. Adhikary, RSC Advances Solvent assisted and solvent free orientation of growth of nanoscaled lanthanide sulfides: tuning of morphology and manifestation of photocatalytic, RSC Adv., 2015, 3, 102818-102827.
- 35 X. Luo, M. Zhang, L. Ma and Y. Peng, Preparation and stabilization of  $\gamma$ -La2S3 at low temperature, J. Rare Earths, 2011, 29, 313-316.
- 36 S. Rezania, A. Mojiri, J. Park, N. Nawrot, E. Wojciechowska, N. Marraiki and N. S. S. Zaghloul, Removal of lead ions from

- wastewater using lanthanum sulfide nanoparticle decorated over magnetic graphene oxide, Environ. Res., 2022, 204, 111959.
- 37 L. Li, S. Hirai and Y. Tasaki, Synthesis and sintering of samarium rich SmSx and its electrical property, I. Rare Earths, 2016, 34, 1042-1047.
- 38 K. R. Kolte, K. Shah and B. S. Chakrabarty, Exploring lanthanum sulphide characteristics for its physical properties, Int. J. Sci. Technol. Res., 2019, 8, 957-961.
- 39 G. Avashthi and M. Singh, Ultrasound accelerated near-edge functionalized heterogeneous graphene oxide sonocatalyst for surface optical bandwidth efficacy and in situ sonothermocatalysis, New J. Chem., 2021, 45, 5463-5483.
- 40 S. Dev and M. Singh, Lanthanide sulfide decorated graphene oxide nanorods as an effective photocatalysts for reducing dyes, free radicals, and water split to produce H2 in sunlight, Solid State Sci., 2022, 134, 107045.
- 41 N. Vashistha, A. Chandra and M. Singh, HSA functionalized Gd<sub>2</sub>O<sub>3</sub>:Eu<sup>3+</sup> nanoparticles as an MRI contrast agent and a potential luminescent probe for Fe<sup>3+</sup>, Cr<sup>3+</sup>, and Cu<sup>2+</sup> detection in water, New J. Chem., 2020, 44, 14211-14227.
- S. S. Maktedar, S. S. Mehetre, G. Avashthi and M. Singh, In situ sonochemical reduction and direct functionalization of graphene oxide: A robust approach with thermal and biomedical applications, Ultrason. Sonochem., 2017, 34, 67-77.
- 43 P. Aryanrad, H. R. Naderi, E. Kohan, M. R. Ganjali, M. Baghernejad and A. Shiralizadeh Dezfuli, Europium oxide nanorod-reduced graphene oxide nanocomposites towards supercapacitors, RSC Adv., 2020, 10, 17543-17551.
- 44 B. Hou, X. Wang, J. Wang, J. Yao, H. Zhang, W. Yu, G. Liu, X. Dong and L. Wang, In situ synthesis of homogeneous Ce<sub>2</sub>S<sub>3</sub>/MoS<sub>2</sub> composites and their electrochemical performance for lithium ion batteries, RSC Adv., 2017, 7, 6309-6314.
- M. Singh and S. Singh, Survismeter: Fundamentals, Concepts, and Applications, Jenny Stanford Publishing, 1st edn, 2019, DOI: 10.1201/9780429027611.
- 46 M. Singh, Survismeter-Type I and II for surface tension, viscosity measurements of liquids for academic, and research and development studies, J. Biochem. Biophys. Methods, 2006, 67, 151-161.
- M. Singh, Upper critical solution temperatures for immiscible solvent systems with halide salts, carboxylic acids, surfactants and polynuclear aromatic compounds and benzene derivatives, J. Chem. Thermodyn., 2007, 39, 240-246.
- 48 R. Mirzajani and S. Karimi, Ultrasonic assisted synthesis of magnetic Ni-Ag bimetallic nanoparticles supported on reduced graphene oxide for sonochemical simultaneous removal of sunset yellow and tartrazine dyes by response surface optimization: Application of derivative spectrophot, Ultrason. Sonochem., 2019, 50, 239-250.