

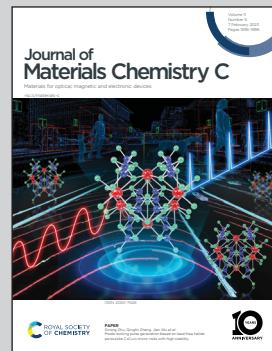
10th anniversary

Showcasing collaborative research from Nara Institute of Science and Technology (NAIST) and National Institute of Advanced Industrial Science (AIST), Japan.

Polymorph- and molecular alignment-dependent lasing behaviors of a cyano-substituted thiophene/phenylene co-oligomer

By changing the crystal preparation method, optically pumped lasing was achieved using four types of crystals with different molecular orientations and crystal morphologies consisting of 5,5'-bis(4'-cyanobiphenyl-4-yl)-2,2'-bithiophene (BP2T-CN).

As featured in:



See Tomomi Jinjyo,
Hitoshi Mizuno *et al.*,
J. Mater. Chem. C, 2023, **11**, 1714.



Cite this: *J. Mater. Chem. C*, 2023,
11, 1714

Polymorph- and molecular alignment-dependent lasing behaviors of a cyano-substituted thiophene/phenylene co-oligomer[†]

Tomomi Jinjyo,^{*a} Hitoshi Mizuno, ^{*a} Fumio Sasaki^b and Hisao Yanagi^a

Four types of crystals with different molecular orientations and crystal morphologies (green platelet, yellow platelet, yellow fiber, and orange disk) consisting of 5,5'-bis(4'-cyanobiphenyl-4-yl)-2,2'-bithiophene (BP2T-CN), a type of thiophene/phenylene co-oligomer (TPCO), were obtained by changing the crystal preparation method. Each crystal exhibited high photoluminescence quantum efficiency values of 61–85%. For green- and yellow-emitting (platelet and fiber) crystals, lasing was observed from the crystal end facets acting as Fabry-Pérot (F-P) resonators. Strong exciton–photon coupling was demonstrated by energy-wavevector dispersion plots obtained from the energies of each interference peak in the mode structure of the F-P resonator. Moreover, a low lasing threshold of $17 \mu\text{J cm}^{-2}$ was achieved in the case of yellow-emitting fiber-like crystals, owing to the high absorption efficiency resulting from the lying-down molecular orientation in the crystal. For the orange-emitting crystal, whispering-gallery mode lasing was observed above $156 \mu\text{J cm}^{-2}$. This higher threshold resulted from the low absorption efficiency that is attributable to the 50° angle (slip angle) formed between the transition dipole moment and crystal basal plane ((100) plane). The crystal structure- and molecular alignment-dependent optical properties can be obtained by controlling the molecular packing and magnitude of intermolecular interactions.

Received 30th September 2022,
Accepted 21st December 2022

DOI: 10.1039/d2tc04151k

rsc.li/materials-c

1. Introduction

The solid-state fluorescence of organic molecules strongly depends on their molecular structure and intermolecular interactions.^{1–14} It is possible to achieve conformation-dependent optoelectronic properties and varied emission colors from the same fluorophore using crystal polymorphs, which can be controlled by preparation methods.^{1–14} The use of molecular alignment-control technology allows us to readily change the crystal morphology from one to three dimensions and depending on the structure of the freely constructed molecular array, different optoelectronic properties are manifested (conformation-dependent optoelectronic properties). Since low-dimensional organic crystals have well-defined crystal planes wherein the molecules are aligned in a regular fashion with specific orientations, the molecular arrangement in a particular crystal plane affects the overall photoluminescence (PL) properties and photoluminescence quantum yield (PLQY).^{15,16} In pentacene

crystals, crystal-plane-dependent PL behavior has been observed. The PL properties have been attributed to the arrangement of pentacene molecules in the (010) and (001) planes and the main crystal planes in one-dimensional (wire) and two-dimensional (disk) crystals.^{15,16} The difference in the PL spectra of one-dimensional wires and two-dimensional disks has been ascribed to the alignment of pentacene molecules on each crystal plane relative to the direction of incident light.^{15,16} Moreover, the molecular alignment is generally governed by the interaction between molecules and substrates, base layer (buffer layer) material, and molecular modification. In organic light-emitting field-effect transistors,^{18–23} organic light-emitting diodes,^{18,22,24,25} organic photovoltaic cells,^{26–28} and organic solid-state lasers,^{18,29–37} achieving a lying or upright molecular orientation with respect to the substrate is important for improving the device performance particularly in terms of carrier transport properties. The use of crystal polymorphs is an efficient molecular alignment-control approach, including the interaction between the molecules and substrates and the use of a buffer layer.^{1–14} Using the crystal polymorphs, the optoelectronic properties can be varied without considering the substrate or buffer layer.

Crystal polymorphs have frequently been used to tune the singlet emissions of organic dyes. However, crystal polymorph-based organic solid-state lasers are under-developed owing to the lack of investigation on the relationship between molecular

^a Graduate School of Science and Technology, Nara Institute of Science and Technology (NAIST), 8916-5 Takayama, Ikoma, Nara 630-0192, Japan.

E-mail: jinjo.tomomi.jn0@ms.naist.jp, hitoshi352-17@ms.naist.jp

^b Research Institute for Advanced Electronics and Photonics, National Institute of Advanced Industrial Science and Technology, Ibaraki 305-8568, Japan

[†] Electronic supplementary information (ESI) available. See DOI: <https://doi.org/10.1039/d2tc04151k>



conformations/structures and optical properties. In a study investigating the impact of polymorphism on the optoelectronic properties of a low-bandgap polymer, two different semi-crystalline polymorphs, β_1 and β_2 , were explored. β_2 has been reported to have a lower optical bandgap, reduced π -stacking distance, higher hole mobility in field-effect transistors, and improved photocurrent generation in polymer solar cells.²⁸ In a study on the effect of crystal polymorphs (consisting of thienothiophene–thiazolothiazole molecules) on the solid-state and transistor properties, Schneider *et al.* reported that among the two crystal polymorphs of 1-Red and 1-Yellow, only 1-Yellow displayed semiconductor properties (p-type, hole mobility (μ_h): $3.3 \times 10^{-3} \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$).¹⁰

For the organic polymorphs of blue- or green-emissive microribbons (MRs) consisting of difluoroboron avobenzone (BF_2AVB), it has been reported that the blue- and green-emissive MRs exhibit monoclinic (J-aggregates, PL efficiency: 68%) and orthorhombic phases (H-aggregates, PL efficiency: 24%), respectively.⁶ The laser oscillation thresholds for the blue- and green-emissive MRs (observed with femtosecond laser pumping) were as high as 530 and $1126 \mu\text{J cm}^{-2}$. Fluorene trimer 2,2':7',2''-ter(9,9-dimethylfluorene) (TDMeF) has four polymorphs with different molecular conformations.⁷ However, all the crystalline polymorphs of TDMeF exhibited blue emissions, and no color tuning was observed.⁷ For the lasing behavior of TDMeF crystals, the quality factor (Q) estimated from the lasing spectra was as low as 786 or 1144.

Meanwhile, the occurrence of aggregation-induced emission and aggregation quenching (concentration quenching) has been reported in many organic crystals and aggregates.¹⁴ Among various organic crystals, single crystals of thiophene/phenylene co-oligomers (TPCOs)^{21,34–48} exhibit high PLQYs ranging from 40 to 63% even in the solid state. To develop high-performance organic lasers and organic light-emitting devices, materials that exhibit high PLQYs, irrespective of conformational and molecular packing changes, are required. The formation and optical properties of crystal polymorphs of TPCO crystals have not been reported yet. Herein, we report the optical properties of crystal polymorphs with controlled crystal structures, molecular orientations, and crystal growth planes using 5,5'-bis(4'-cyanobiphenyl-4-yl)-2,2'-bithiophene (BP2T-CN) (n-type),^{37,38,48} a type of TPCO. We also demonstrate the optically pumped lasing properties of four different BP2T-CN crystals with controlled crystal morphologies and molecular alignments.

2. Experimental section

BP2T-CN was purchased from Sumitomo Seika Chemicals Co. and was used as such, without further purification. Tetrahydrofuran (THF) and *o*-dichlorobenzene (*o*-DCB) were purchased from Sigma-Aldrich Co. Ltd.

BP2T-CN single crystals were prepared using physical vapor transport (PVT) and solution growth methods. In the PVT method, the BP2T-CN powder, placed in a nitrogen-filled glass tube was heated at 280 °C for 24 h using an electric furnace with

a temperature gradient, and green emitting BP2T-CN crystals were obtained. The flow rate of nitrogen gas was adjusted to 0.01 L min⁻¹ using a flow meter. After cooling to 40 °C, the single crystals were selected and mounted onto a quartz substrate using a tungsten needle. When preparing the platelet- or fiber-like BP2T-CN crystals exhibiting yellow emission, solution growth method was used. To grow the platelet-like crystals, 2 mg of BP2T-CN powder was dissolved in 20 mL of THF, and the solution was heated to 60 °C. The single crystals were precipitated by gradual cooling to 27 °C for 24 h using a temperature controller. After cooling, the solution was filtered and allowed to dry naturally. Single crystals on the filter were then selected and mounted onto a quartz substrate using a tungsten needle. To prepare the fiber-like crystals, 3 mg of BP2T-CN powder was dissolved in 20 mL of *o*-DCB, and the solution was heated to 150 °C. The BP2T-CN fiber-like crystals were precipitated after slow cooling to 27 °C for 20 h using a temperature controller. BP2T-CN crystals with orange emission were prepared using the melting method. BP2T-CN powder was placed on a quartz substrate and melted using a gas burner. The heating was stopped immediately after the BP2T-CN powder was melted over a gas burner. The temperature of the quartz substrate surface was determined to be approximately 360 °C by using a thermometer. The prepared BP2T-CN crystals were allowed to cool naturally.

Optical absorption spectra were measured using a UV-visible spectrometer (JASCO Japan Spectroscopy V-730). The fluorescence micrographs of the BP2T-CN crystals were obtained using a fluorescence microscope (Olympus, BX-51). The PL spectra were measured using a spectrometer (Princeton Instruments, HRS-300) equipped with a thermoelectrically cooled CCD (charge-coupled device) detector (Princeton Instruments, PIXIS-256) and a neodymium (Nd):yttrium aluminum garnet (YAG) laser ($\lambda_{\text{ex}} = 355 \text{ nm}$, pulse duration: <1.1 ns, repetition rate: 1.2 kHz). $\theta/2\theta$ measurements of the single crystal were performed at 103 K using an X-ray diffractometer (Rigaku ValiMax Rapid) with a Mo K α line ($\lambda = 0.71075 \text{ \AA}$). For time-resolved PL measurements, the second-harmonic light of a Ti:sapphire pulsed laser (Coherent Mira 900F, time duration: 100–150 fs, repetition rate: 76 MHz) was used as an excitation source ($\lambda_{\text{ex}} = 400 \text{ nm}$). The luminescence of the single crystals was measured and focused on the entrance slit of an imaging polychromator (Hamamatsu Photonics, C5094). The emission decay profiles of the BP2T-CN crystals were measured using a streak camera (Hamamatsu, C4334) (time resolution: 15 ps).

Differential scanning calorimetry (DSC) purity analysis and thermogravimetry-differential thermal analysis (TG-DTA) were performed on a Hitachi DSC7000X/STA7200 under nitrogen gas flow at a heating rate of 1 °C min⁻¹ and 20 °C min⁻¹, respectively.

3. Results and discussion

The molecular structure of BP2T-CN is shown in Fig. 1(a). Fig. 1(b) and (c) show the fluorescence micrographs of the



BP2T-CN single crystals prepared using the THF solvent by the PVT and liquid-phase growth methods, respectively. The BP2T-CN single crystals prepared by the PVT method showed green emission, and bright emission was observed at the crystal edges. On the other hand, BP2T-CN single crystals grown using the liquid-phase growth method showed yellow emission from the entire crystal body. The PL and absorption spectra of the green- and yellow-emitting crystals, measured at room temperature (300 K) are shown in Fig. 1(d) and (e), respectively. For the green-emitting crystals prepared by the PVT method (Fig. 1(d)), the 0-1, 0-2, and 0-3 PL bands were observed at 2.34, 2.17, and 2.00 eV, respectively. The 0-0, 0-1, and 0-2 absorption bands were observed at 2.56, 2.73, and 2.90 eV, respectively. In the yellow-emitting crystals prepared using the liquid-phase growth method (Fig. 1(e)), the 0-1, 0-2, and 0-3 transitions in the PL spectra appear at 2.26, 2.09, and 1.92 eV, respectively.

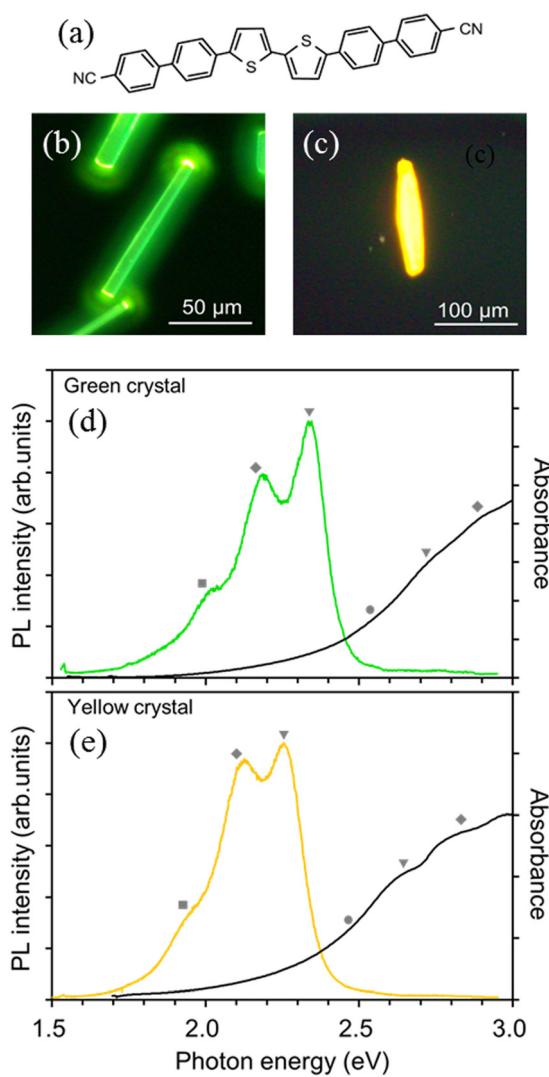


Fig. 1 Molecular structure of BP2T-CN (a). Fluorescence micrographs of green- (b) and yellow-emitting (c) BP2T-CN crystals. Optical absorption and photoluminescence (PL) spectra of the green- (d) and yellow-emitting (e) crystals. Gray circle, triangles, diamonds, and squares in the absorption and PL spectra indicate the 0-0, 0-1, 0-2, and 0-3 bands, respectively.

The 0-0, 0-1, and 0-2 absorption bands appear at 2.48, 2.65, and 2.82 eV. Although the lowest 0-0 transition is forbidden, weak 0-0 transitions in the absorption spectra of BP2T-CN crystals (similar to other TPCO crystals) were observed which can be attributed to the antiparallel exciton coupling between adjacent molecules aligned in the crystal. Compared to the energy of the 0-1 PL bands of the green-emitting single crystal, the yellow-emitting crystal was red-shifted by 0.07 eV.

X-ray structural analysis of the BP2T-CN crystals was performed to investigate the factors responsible for the energy difference in the PL spectra of the green- and yellow-emitting crystals. The crystal structures of the green- and yellow-emitting crystals are shown in Fig. 2(a)–(d). The crystal structure of the green-emitting crystal prepared by the PVT method was reproduced using the data reported by Hatano *et al.* (monoclinic, $a = 18.34 \text{ \AA}$, $b = 7.24 \text{ \AA}$, $c = 18.446 \text{ \AA}$, $\beta = 100.5^\circ$).³⁷ The transition dipole moment of the BP2T-CN molecule is parallel to the long molecular axis, and the dipole moments (molecules) align parallel to the ac plane of the crystal, as shown in Fig. 2(a). The crystal plane of the green-emitting crystal is (201), and the long axis of the molecule forms an angle of 25° against the crystal plane, as shown in Fig. 2(a). Because of the lying orientation (with respect to the crystal plane) of the molecules, both surface and edge emissions were obtained from the platelet-like crystals (Fig. 1(b)). As shown in Fig. 2(b), the shortest distance between two adjacent molecules was 6.41 \AA . Single-crystal X-ray structure analysis of the yellow-emitting

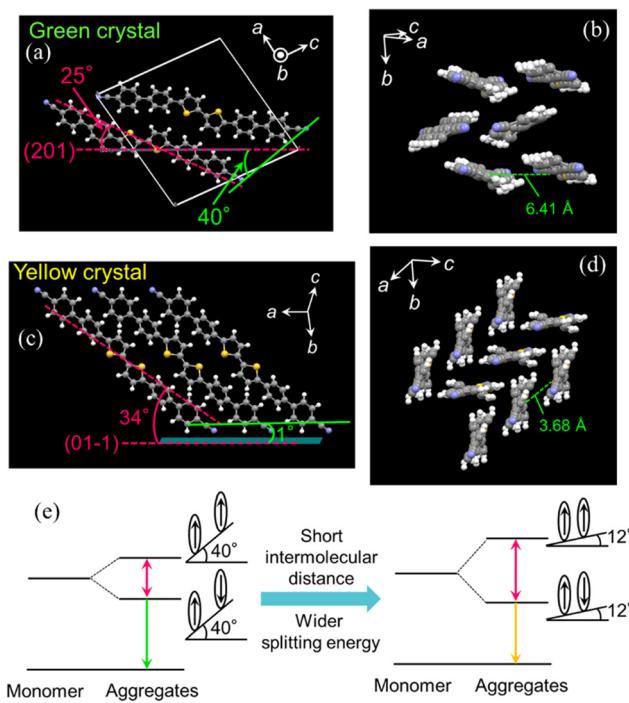


Fig. 2 Crystal structures of the green- (a and b) and yellow-emitting (c and d) BP2T-CN crystals. The crystal structures of the green-emitting crystal were reproduced using the reported data.³⁷ Schematic diagram for the splitting energy in molecular aggregates increasing with decreasing intermolecular distance and slip angle (e). Slip angles are also shown in (e).

crystals shown in Fig. 1(c) revealed that the BP2T-CN molecules formed a triclinic crystal with $a = 7.124 \text{ \AA}$, $b = 12.255 \text{ \AA}$, $c = 14.512 \text{ \AA}$, $\alpha = 98.39^\circ$, $\beta = 102.08^\circ$, and $\gamma = 97.67^\circ$. Thus, BP2T-CN forms two different molecular arrangements and crystal structures, green- and yellow-emitting crystals as crystalline polymorphs. The differences in molecular packing change the magnitude of intermolecular interactions, thereby changing the optical transition energy. In case of the yellow-emitting crystal, the BP2T-CN molecules align along the ac plane of the crystal because the crystal plane of the platelet-like crystal structure (Fig. 1(c)) corresponds to the (01-1) plane. The molecular long axis forms an angle of 34° against the crystal basal plane ((01-1) plane), and the molecules result in an inclined orientations to the basal plane. The closest distance between the two neighboring molecules is 3.68 \AA (Fig. 2(d)), which is approximately 0.57 times lesser than that in the case of green-emitting crystals. The mechanism responsible for the energy change in response to the change in molecular arrangement and crystal structure is explained as follows: With the aggregation of organic molecules, the excited energy levels split into two levels (high and low energy levels), as shown in Fig. 2(e).¹ For a majority of TPCO crystals reported so far, the lowest excited energy level is forbidden due to the antiparallel exciton coupling in the excited state, as discussed above. However, the misaligned molecular arrangements, as in the case of BP2T-CN (Fig. 2(e)) result in a partially allowed transition at the lowest excited energy level.^{31,32,36,45} As the intermolecular distance between the two neighboring molecules shortens, the width of energy splitting increases, and the lowest excited energy level becomes more stable. Therefore, it is inferred that the emission color changed from green to yellow because the distance between the two closest neighboring molecules was shorter and the lowest excited energy level was more stabilized in the yellow polymorphs prepared by the liquid-phase growth method. The dimer model can also explain this change in emission energy. In the dimer model reported by Kasha,⁴⁹ excitonic interaction energy between two transition dipole moments is given by $E_{\text{exciton}} = |\mu|^2(1 - 3\cos^2\theta)/r^3$, where E_{exciton} , μ , θ , and r are excitonic interaction energy, transition dipole moment, angle, and distance between two transition dipole moments, respectively. In the crystal polymorphs of BP2T-CN, the angle and distance between two transition dipole moments affect the excitonic interaction energy. Using the transition dipole moment (16.3 D),⁴⁸ slip angle (angle between two adjacent molecules forming a diagonal pair and crystal basal plane) (40° and 1°) (see Fig. 2(a) and (c)), the distance between the two dipoles (6.41 \AA and 3.68 \AA) (see Fig. 2(a) and (c)), and the magnitude of E_{exciton} are larger in the yellow-emitting crystal than that in the green-emitting crystal. Hence, since the splitting energy in the dimer is larger in the yellow-emitting crystal than that in the green-emitting crystal, the red-shifted emission is obtained in the yellow-emitting crystal.

To investigate the emission lifetimes and PLQYs of the green and yellow crystal polymorphs, we measured the temperature dependence of their time-resolved PL spectra (10–300 K). Fig. 3(a) and (b) show the decay curves of the 0-1 emission

bands measured at 300 K for the green and yellow crystal polymorphs, respectively. Using convolution fitting (sky blue solid line) with the incident laser pulse (black dashed line), the emission decay profiles were characterized by two time constants: 530 ps (89%) and 1.3 ns (11%) for the green-emitting crystal, and 720 ps (99%) and 2.5 ns (1%) in the case of the yellow-emitting crystal. In the green crystal, BP2T-CN platelet crystal polymorphs, the fast and slow components are attributed to the two different types of singlet exciton states: free and self-trapped excitons (shallow self-trapped excitons), respectively, as reported earlier.⁴⁸ It can be inferred from this mechanism that the main components with 530 ps time constant in the green crystal polymorph and 720 ps time constant in the yellow crystal polymorph are due to the free excitons. To determine the PLQY values for crystal polymorphs, the temperature dependence of emission lifetimes of 0-1 band (dominant component) was analyzed using the following equation, which has been used in earlier works.^{43,48}

$$\tau(T) = 1/(k_f + k_{nr} \exp(-E_a/(k_B T))) \quad (1)$$

where τ , k_f , k_{nr} , E_a , k_B , and T are the spontaneous emission lifetime, radiative transition rate constant, nonradiative transition rate constant, activation energy of nonradiative transition, Boltzmann constant, and temperature, respectively. Fig. 3(c) shows the temperature dependence of emission lifetimes of the 0-1 band in the green and yellow crystal polymorphs. The parameters k_f , k_{nr} , and E_a for the green and yellow crystal polymorphs were determined by fitting the experimental data to eqn (1). The values are summarized in Table 1. Using the parameters listed in Table 1, the PLQY(Φ) value was estimated using the equation,

$$\Phi = k_f/(k_f + k_{nr}). \quad (2)$$

The PLQY values are presented in Table 1. In comparison to BP2T-CN green-emitting crystals, yellow-emitting crystals have longer lifetimes (τ) (main component) and higher PLQY and E_a values because of the lower value of k_{nr} . For the green-emitting crystals, a PLQY value of 63% has been reported for BP2T-CN platelet crystals which is almost consistent with the PLQY value (61%) obtained in one of our previous studies.⁴⁸ In the yellow-emitting crystals, a higher PLQY value of 85% was obtained due to the lower k_{nr} and higher E_a values. Meanwhile, in the green-emitting crystals with a larger slip angle of 40° (see Fig. 2(a)), the PLQY value was lower than that of the yellow-emitting crystals because of the larger k_{nr} and lower E_a values. In the crystal of BP2T which is a molecule similar to BP2T-CN, luminescence has been reported to originate from extrinsic origins such as molecular aggregates or defects in the crystal.⁵⁰ The large k_{nr} in the green-emitting crystal with larger slip angle indicating weak intermolecular interactions may be due to the fact that the nonradiative transition rate constants to energy levels originating from molecular aggregates and defects in the crystals are different for green- and yellow-emitting crystals.

To investigate the optically pumped lasing behavior of the green and yellow crystal polymorphs exhibiting high PLQY values, we measured the PL spectra under high-density



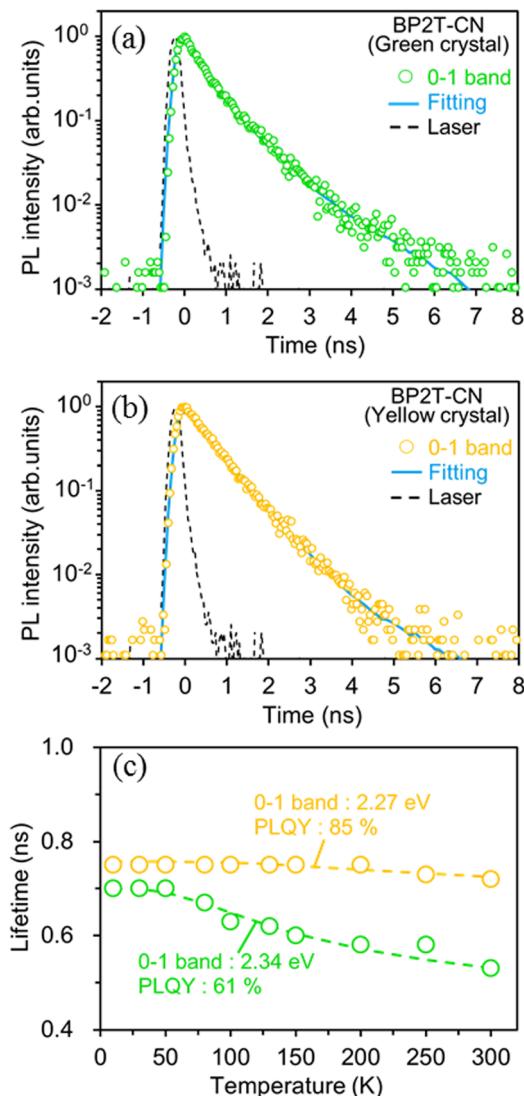


Fig. 3 Emission decay profiles of 0-1 emission bands measured at 300 K for the green (a) and yellow (b) crystal polymorphs. Temperature dependence of the emission lifetimes (dominant component) of the 0-1 bands for the green (green color) and yellow (yellow color) polymorphs (c).

photoexcitation using a Nd:YAG laser. A spot-shaped excitation beam (0.12 cm^2 , p-polarization) was focused on the crystal polymorphs at an incidence angle of 45° against the substrate, and the light emitted from the crystal edges was detected using a CCD spectrometer (Hamamatsu PMA-50). The BP2T-CN green crystal polymorph shown in Fig. 1(b) had a length (L) of $100 \mu\text{m}$ and a thickness (t) of $3.7 \mu\text{m}$, and BP2T-CN yellow crystal

polymorph shown in Fig. 1(c) had L of $120 \mu\text{m}$ and t of $14.8 \mu\text{m}$. Fig. 4(a) and (b) show the excitation density dependence of the PL spectra and integrated PL intensity of the 0-1 band (2.31–2.33 eV) for the green-emitting crystal. For the green-emitting crystals, when the excitation density increased above $69 \mu\text{J cm}^{-2}$, emission amplification was observed in the 0-1 band (2.32 eV). As shown in Fig. 4(b), the integrated 0-1 band intensity increased linearly in the low excitation density range, and super linearly above the threshold of $69 \mu\text{J cm}^{-2}$. In the yellow-emitting single crystal shown in Fig. 1(c), a broad spontaneous emission spectrum was observed at a low excitation density of $46 \mu\text{J cm}^{-2}$ (Fig. 4(c)). The integrated 0-2 band PL intensity (2.10–2.14 eV) increased super linearly at excitation densities above $214 \mu\text{J cm}^{-2}$, as shown in Fig. 4(c) and (d). The emission amplification in the 0-2 band in the yellow-emitting crystal can be ascribed to the reabsorption effect due to the thicker crystal ($14.8 \mu\text{m}$). The lasing thresholds for green- and yellow-emitting crystals are affected by the absorption efficiency of the excitation light owing to their specific molecular orientations with respect to the crystal basal planes. The angles between the electric field plane of the excitation light and the molecular transition dipole moment, against the crystal basal planes ((201) or (01–1) planes) in the green- and yellow-emitting crystals were 110° and 101° , respectively. The yellow-emitting crystal can absorb the excitation light more efficiently than the green-emitting crystal. However, a lower lasing threshold was obtained for the BP2T-CN green-emitting crystal. The one-dimensional optical confinement effect is responsible for a lower threshold lasing in the BP2T-CN green-emitting crystal. Fig. 5(a) and (b) show high-resolution PL spectra showing lasing for the green- and yellow-emitting crystals measured at a spectral resolution of 0.2–0.3 nm. When the excitation density increased above the threshold, longitudinal multimode lasing was observed in the 0-1 and 0-2 emission bands for the green- and yellow-emitting crystals, respectively. With a mode spacing of $\Delta\nu = 11 \text{ cm}^{-1}$ in the lasing spectrum (Fig. 5(a)) and the crystal length L of $100 \mu\text{m}$ (Fig. 1(b)), the group refractive index, n_g of the BP2T-CN green-emitting crystal was estimated to be $n_g = 1/(2 L \Delta\nu) = 4.6$. This n_g value is almost consistent with the values reported for other TPCO crystals.^{34–36} Thus, in the green-emitting crystal shown in Fig. 5(a), the lasing has been attributed to Fabry-Pérot (F-P) resonance effect caused by the longitudinal end faces in the crystal acting as a mirror. The Q value estimated from the lasing spectrum was 4250. For the lasing spectrum in the yellow-emitting crystal shown in Fig. 5(b), n_g was calculated to be 3.2 with a mode spacing of $\Delta\nu = 13 \text{ cm}^{-1}$ in the lasing spectrum and a crystal length L of

Table 1 The parameters obtained from the time-resolved PL measurements for green platelet, yellow platelet, yellow fiber, and orange disk crystals

	Green platelet crystals	Yellow platelet crystals	Yellow fiber crystals	Orange disk crystals
Lifetime (τ) at 300 K	0.53 ns (89%) 1.3 ns (11%)	0.72 ns (99%) 2.5 ns (1%)	0.74 ns	1.1 ns
k_f	$14.3 \times 10^8 \text{ s}^{-1}$	$13.2 \times 10^8 \text{ s}^{-1}$	$11.0 \times 10^8 \text{ s}^{-1}$	$7.6 \times 10^8 \text{ s}^{-1}$
k_{nr}	$9.1 \times 10^8 \text{ s}^{-1}$	$2.3 \times 10^8 \text{ s}^{-1}$	$5.2 \times 10^8 \text{ s}^{-1}$	$4.2 \times 10^8 \text{ s}^{-1}$
E_a	18 meV	35 meV	20 meV	25 meV
PLQY	0.61	0.85	0.68	0.64

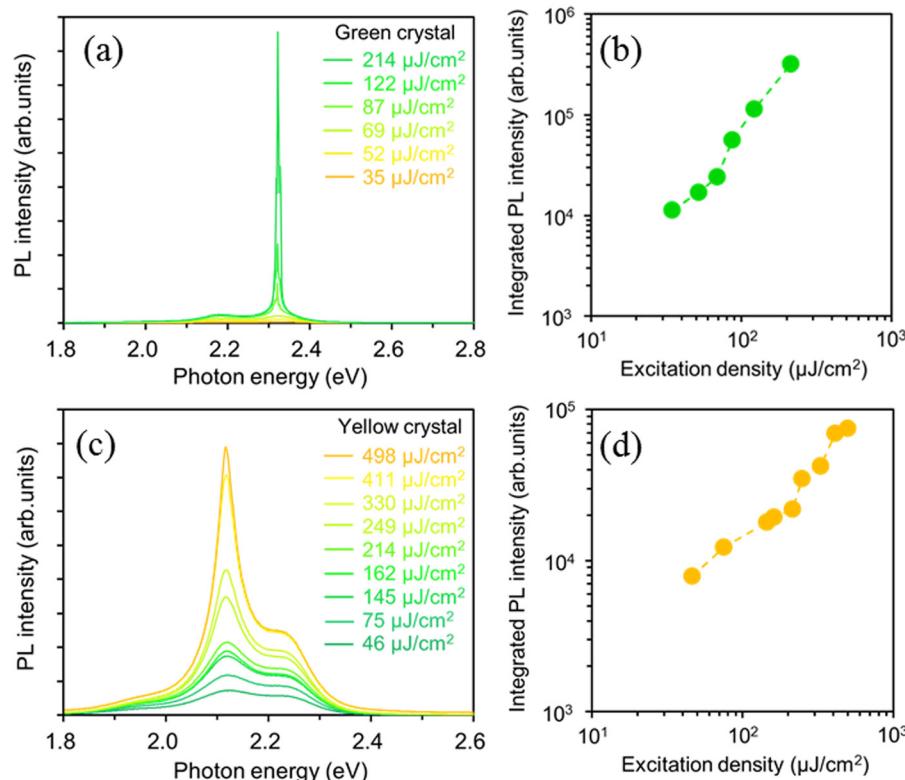


Fig. 4 Excitation density dependence of PL spectra (a) and integrated PL intensity (b) of 0-1 band (2.31–2.33 eV) measured for the green-emitting crystal. Excitation fluence dependence of PL spectra (c) and integrated PL intensity (d) of 0-2 band (2.10–2.14 eV) for the yellow-emitting crystal.

120 μm in the longitudinal direction. Longitudinal multimode lasing was observed in the yellow-emitting crystal, originating from the F-P resonance in the BP2T-CN platelet-like crystal resonator with mirror surfaces at both ends. The Q value for the yellow-emitting crystal, estimated from the lasing spectrum (Fig. 5(b)) was 2360. The lower Q value in the yellow-emitting crystal in comparison to that in the green-emitting crystal can be ascribed to emission leakage from the crystal surface, as shown in Fig. 1(c). The high n_g and Q values obtained in the green-emitting crystal can be attributed to the one-dimensional optical confinement resulting from the well-aligned molecules in the longitudinal direction coupled with the excellent optical waveguide quality.

Next, we show the optical properties of BP2T-CN fiber-like crystals for different growth planes obtained by changing the solvent used in the liquid-phase growth method from THF to *o*-DCB. Fig. 6(a) and (b) show the fluorescence micrograph and PL spectrum of the yellow-emitting fiber-like crystals. In Fig. 6(b), the 0-1, 0-2, and 0-3 transitions in the PL spectrum were observed at 2.30, 2.13, and 1.96 eV, respectively. To examine the molecular orientation of these fiber-like crystals, X-ray diffraction (XRD) $\theta/2\theta$ measurements were performed by mounting the crystals on the glass substrate. The XRD patterns of the fiber-like crystals are shown in Fig. 6(c). Diffraction peaks were observed at 7.36° , 20.37° , and 21.34° , corresponding to the (010), (10-3), and (022) planes, respectively. The crystal structures displaying the (010), (10-3), and (022) planes corresponding to

each crystal basal plane are shown in Fig. 6(d)–(f), respectively. The highest diffraction intensity in the (022) plane suggests that the (022) plane is parallel to the substrate in most fiber-like crystals. The simulated powder XRD pattern (Fig. 6(c)) using the crystallographic data for the yellow-emitting crystal shown in Fig. 1(c) is in good agreement with the diffraction pattern of the fiber-like crystals. This result indicates that the yellow-emitting fiber-like crystal has the same crystal structure as the yellow-emitting platelet-like crystal. In the yellow-emitting fiber-like crystals, the transition dipole moments of the BP2T-CN molecules were tilted by 6° with respect to the (022) plane, and fiber-like crystals were obtained due to π -stacking along the a -axis.

We also obtained the PLQY value by measuring the temperature dependence of emission lifetimes for the BP2T-CN fiber-like crystal using eqn (1). Fig. 6(g) shows the temperature dependence of the 0-1 band emission lifetimes for the fiber-like crystal. The emission decay profile exhibited a single exponential decay with a lifetime of 0.74 ns at 300 K. The parameters were listed in Table 1.

We examined the effects of the molecular orientation and crystal shape on the lasing properties. The fiber-like crystal shown in Fig. 6(a) had L of 40 μm , crystal width of approximately 1.6 μm , and t of 100 nm. As shown in Fig. 7(a) and (b), as the excitation density increased above $17 \mu\text{J cm}^{-2}$, the PL intensity increased significantly with a decrease in the emission spectral width. Fig. 7(c) shows the high-resolution PL spectrum (spectral resolution: 0.2–0.3 nm) of the fiber-like crystal at



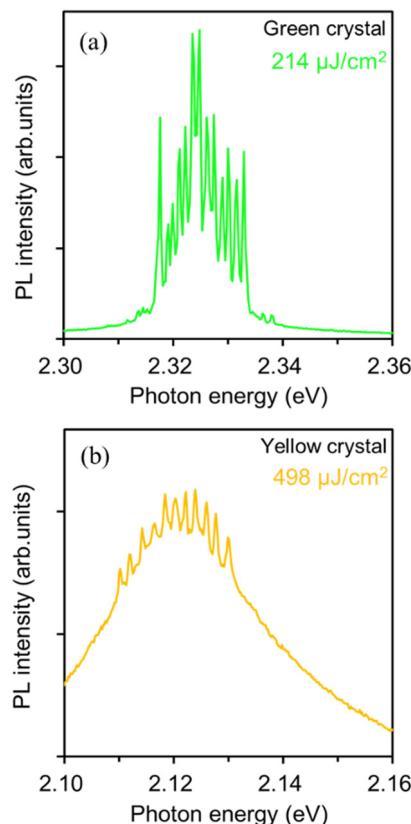


Fig. 5 High-resolution PL spectra (spectral resolution: 0.2–0.3 nm) showing lasing for the green- (a) and yellow-emitting (b) crystals.

85 $\mu\text{J}/\text{cm}^2$. As shown in Fig. 7(b), when the excitation density increased above the threshold excitation density of 17 $\mu\text{J}/\text{cm}^2$, longitudinal multimode lasing was observed in the 0-2 emission band. The lower lasing threshold of 17 $\mu\text{J}/\text{cm}^2$ can be attributed to the two-dimensional optical confinement along the fiber direction and the higher absorption efficiency to the lying molecular orientation (molecules were tilted by 6° against the (022) plane). Despite the thin thickness of 100 nm, lasing was observed in the 0-2 emission band probably due to the reabsorption effect. The reabsorption effect is ascribed to the increased absorption efficiency resulting from more lying-down molecular orientations in comparison to the yellow-emitting platelet-like crystal (Fig. 1(c)). The n_g value of the fiber-like crystal was determined to be 7.8 with mode spacing, $\Delta\nu$ of 16 cm^{-1} and L of 40 μm as shown in the lasing spectrum in Fig. 7(c). The Q value estimated from the lasing spectrum was 3700. The higher n_g value in the fiber-like crystal compared to the green-emitting platelet-like and the yellow-emitting platelet-like crystal may be due to the fact that the emission propagates as exciton polariton along the fiber-like crystal.^{48,51} We have already reported that the lower polariton branch propagates along the BP2T-CN microrod crystal in our previous work.⁴⁸ The exciton-polariton property in the BP2T-CN fiber-like crystal is described below. Fig. 8(a) shows a schematic representation of exciton-polariton formation in BP2T-CN fiber-like crystal. Fig. 8(b) represents the PL spectrum of BP2T-CN

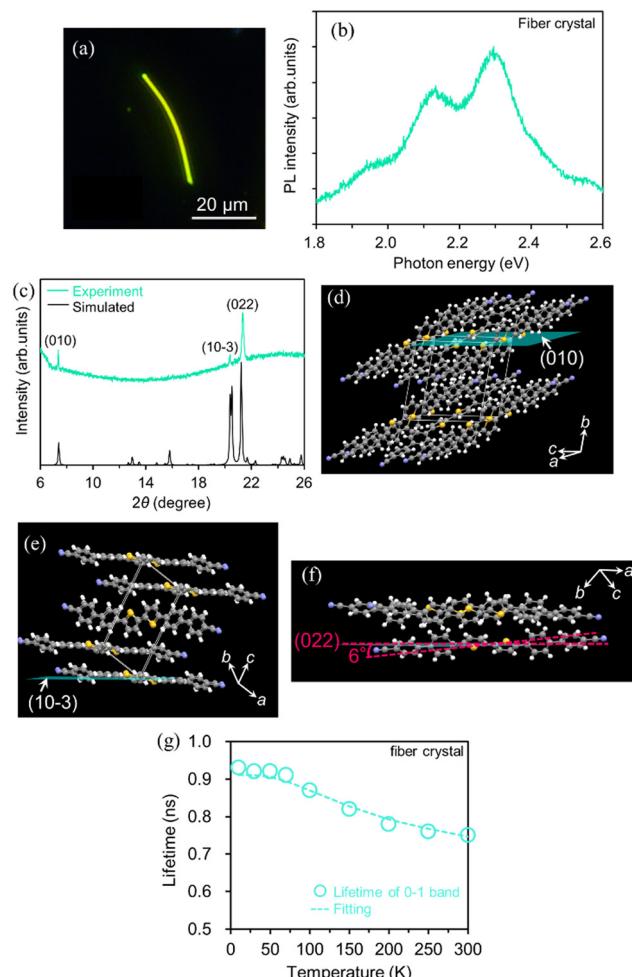


Fig. 6 Fluorescence image (a) and fluorescence spectrum (b) of BP2T-CN fiber-like crystal. X-ray diffraction (XRD) pattern (green) measured using the fiber-like crystals mounted on the glass substrate (c). The simulated powder XRD pattern using the crystallographic data for the yellow-emitting crystal was also shown by a black line in (c). Crystal structures displaying (010) (d), (10-3) (e), and (022) (f) planes corresponding to each crystal basal plane. (g) The temperature dependence of the 0-1 band emission lifetimes for the fiber-like crystal.

fiber-like crystal detected from the crystal edge (blue line). In Fig. 8(b), the blue solid, red solid, and black dashed lines indicate the PL spectrum, reproduced spectrum using the sum of Lorentzian functions, and Lorentz functions corresponding to each resonance energy, respectively. Using the PL spectrum in Fig. 8(b), the corresponding energy-wavevector ($E-k_z$) dispersion of propagating modes was obtained by plotting the energies of each interference peak (energy of each Lorentzian function) with respect to k_z . The $E-k_z$ dispersion plots are shown in Fig. 8(c). To demonstrate exciton-polariton formation in the BP2T-CN fiber-like crystal, the exciton-polariton dispersion was analyzed using a coupled oscillator model for exciton-photon coupling:⁴⁸

$$E = \frac{1}{2}(E_{\text{ex}} + E_{\text{ph}}) - \frac{1}{2}\sqrt{(E_{\text{ex}} - E_{\text{ph}})^2 + \Omega^2} \quad (3)$$



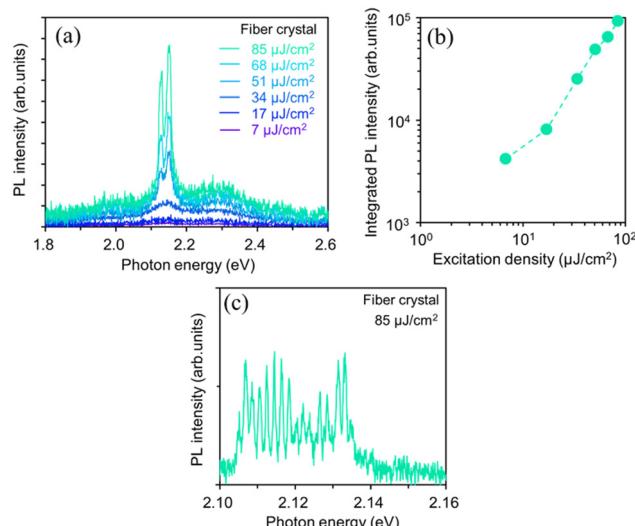


Fig. 7 PL spectra (a) and integrated 0-2 emission band intensity (b) of BP2T-CN fiber-like crystal measured as a function of excitation density. High-resolution PL spectrum (spectral resolution: 0.2–0.3 nm) measured at $85 \mu\text{J cm}^{-2}$ in the fiber-like crystal (c).

where E_{ex} , E_{ph} , and Ω are the energies of the bare exciton, uncoupled cavity photon, and Rabi splitting, respectively. The cavity photon dispersion E_{ph} is expressed by the following equation:⁴⁸

$$E_{\text{ph}} = \frac{c}{n} \sqrt{\hbar^2 k^2 + m_0^2 \left(\frac{c}{n}\right)^2} \quad (4)$$

where c is the speed of light, n is the refractive index, k ($= 2\pi n/\lambda$) is the wavevector of the exciton polariton, \hbar is Planck's constant,

and m_0 is the cavity photon mass determined by the optical confinement imposed by the cross-section.⁴⁸ The experimental data plots showed good agreement with the coupled oscillator model for exciton–photon coupling (eqn (3)). We determined the fitting parameters as E_{ex} (0-0 transition energy estimated from vibrational spacing of 0.17 eV in the PL spectrum in Fig. 6(b)) = 2.47 eV, $n = 4.8$, $m_0 = 9.35 \times 10^{-17} \text{ eV s}^2 \text{ m}^{-2}$, and $\Omega = 1.3$ eV. The Rabi splitting energies in organic/CH₃NH₃PbBr₃ nanowires and organic microresonators have been reported to be 0.564–1.09 eV.^{52–54} The Rabi splitting energy of 1.3 eV obtained in the present study has comparable value with that of our earlier report (Ω : 1.25 eV).⁴⁸

Next, we discuss the emission properties of orange-emitting crystals with molecular orientations different from the three BP2T-CN crystals described above. Fig. 9(a) and (b) show the fluorescence image, absorption, and PL spectra of the BP2T-CN orange-emitting crystal prepared using the melting method. Polarized light microscopy observations confirmed that orange-emitting samples were single crystals (see the ESI,† Fig. S1). The 0-0, 0-1, 0-2, and 0-3 absorption bands appeared at 2.41, 2.59, 2.77, and 2.95 eV, respectively. The 0-1, 0-2, and 0-3 PL bands are observed at 2.22, 2.04, and 1.86 eV, respectively. XRD measurements were performed to investigate the molecular orientation of the BP2T-CN orange-emitting crystals. The results demonstrated that the crystal structure of the orange-emitting BP2T-CN crystals was the same as that of the green-emitting crystal (Fig. 1(b)) (Fig. S2, ESI†). The orange luminescence was not caused by impurities or the mixture of amorphous and monoclinic phase crystal. To investigate that the orange emission was not due to thermal decomposition

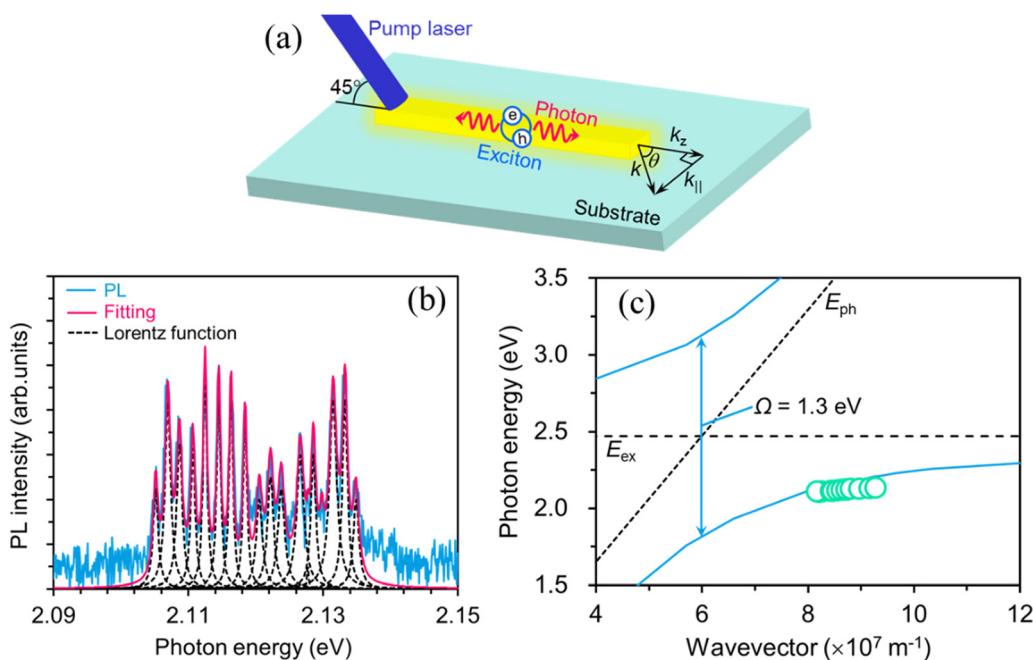


Fig. 8 (a) Schematic representation of exciton–polariton formation in BP2T-CN fiber-like crystal. (b) PL spectrum of BP2T-CN fiber-like crystal detected from the crystal edge (blue line). The red line shows reproduced spectrum using the sum of Lorentzian functions corresponding to each resonance energy (black dashed lines). (c) Energy–wavevector ($E-k_z$) dispersion plots obtained from the energies of each interference peak (energy of each Lorentzian function) in (b). The blue lines in (c) show upper and lower polariton branches (UPB, LPB) analyzed using a coupled oscillator model.

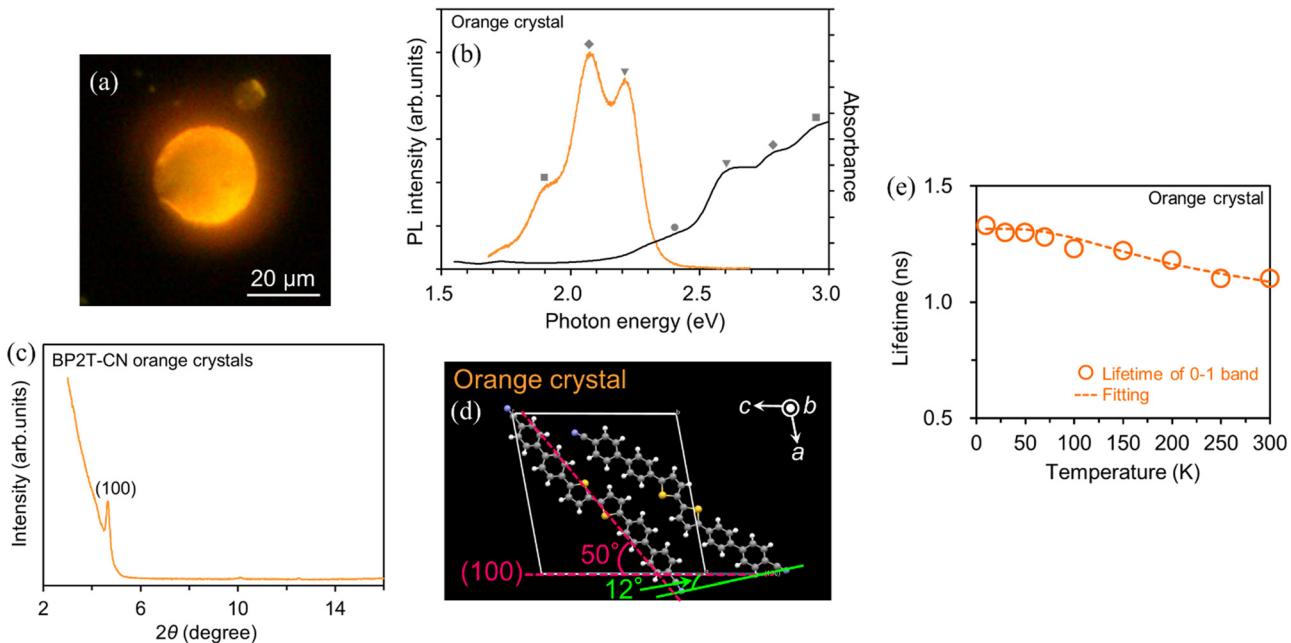


Fig. 9 Fluorescence image (a), absorption (black), and PL spectra (orange) (b) of the BP2T-CN orange-emitting crystal. Gray circle, triangles, diamonds, and squares in the absorption and PL spectra indicate the 0-0, 0-1, 0-2, and 0-3 bands, respectively. XRD patterns measured for the orange-emitting crystals (c). Crystal structure of the orange-emitting crystal displaying a (100) plane (d). (e) The temperature dependence of the emission lifetimes of the 0-1 band for the orange crystal.

or impurities, the DSC purity analysis and TG-DTA were performed. The DSC purity analysis resulted in the purity of 99.64% for the BP2T-CN orange-emitting crystals. From the TG-DTA spectra, it was found that no thermal decomposition did not occur under the crystal fabrication condition at approximately 360 °C using a gas burner (Fig. S3, ESI†). The difference in the peak energy in the PL spectra between the orange- and green-emitting crystals is on account of the crystal plane-dependence of the PL properties. Few studies have been conducted on such crystal plane-dependent PL properties.^{15,16} It is conceivable that the emission energy was red-shifted in the orange-emitting crystal because of the appearance of the (100) plane (basal plane), not observed in the case of the green-emitting crystal. Meanwhile, in 9,10-bis(phenylethynyl)-anthracene (BPEA) microtubes/microrods, it has been reported that the optical loss in the BPEA microtubes is ascribed to reabsorption, while scattering and the coupling between the BPEA microrod crystal and the substrate that causes optical loss (waveguide loss) in light propagation are responsible for the change in emission color.^{16,17} Because the orange-emitting crystal retains its emission color even after being peeled from the substrate, it implies that there is negligible coupling between the crystal and the substrate. In addition, in the BP2T-CN orange-emitting crystal, wherein luminescence is confined within the disk-shaped crystal, it is unlikely that optical loss may be the cause of the red-shifted PL spectrum. Therefore, the red-shifting of approximately 0.12 eV in the PL spectrum of the orange-emitting crystal in comparison to the green-emitting crystal is attributed to the crystal plane change.

Fig. 9(c) shows the XRD patterns of the orange-emitting crystals. The diffraction pattern was indexed using the crystallographic data for the green-emitting crystal shown in Fig. 1(b). A diffraction peak, observed at 4.65° (Fig. 9(c)) corresponds to the (100) plane, as shown in Fig. 9(d). For the molecular orientation in the orange-emitting crystal, the angle between the transition dipole moment of the BP2T-CN molecule and the (100) plane is 50°, as shown in Fig. 9(d). In the orange-emitting crystal as shown in Fig. 9(d), the slip angle, which is the angle between two adjacent molecules forming a diagonal pair and the (100) plane, was 12°. This smaller slip angle compared to that in the green-emitting crystal is the cause of the red-shifted spectra. Comparison of the parameters obtained from four types of BP2T-CN crystals is summarized in Table 2.

Fig. 9(e) shows the temperature dependence of the emission lifetimes of the 0-1 band for the orange crystal. The PLQY value was determined for the orange-emitting crystal using eqn (1). The emission decay curve showed a single exponential decay with a lifetime of 1.1 ns at 300 K. The obtained parameters are summarized in Table 1.

To investigate the effects of the molecular orientation and crystal geometry on the light amplification properties, we examined the lasing properties of the orange-emitting crystal. The orange-emitting crystal shown in Fig. 9(a) is a disk-shaped crystal with a diameter, R of 29 μm and a thickness, t of 600 nm. Fig. 10(a) and (b) show the excitation-density dependence of the PL spectra and the integrated PL intensity of the orange-emitting crystal. As the excitation density increased above 156 μJ cm⁻², the PL intensity in the 0-2 emission band (2.06 eV) increased super linearly with the increase in excitation



Table 2 Comparison of the parameters obtained from four types of BP2T-CN crystals

	Green platelet (Fig. 1(b))	Yellow platelet (Fig. 1(c))	Yellow fiber (Fig. 6(a))	Orange disk (Fig. 9(a))
Crystal structure	Monoclinic	Triclinic	Triclinic	Monoclinic
Crystal basal plane	(201)	(01-1)	(022)	(100)
Angle between the long axis of molecule and the crystal basal plane	25°	34°	6°	50°
Slip angle	40°	1°		12°
Lasing threshold	69 $\mu\text{J cm}^{-2}$	214 $\mu\text{J cm}^{-2}$	17 $\mu\text{J cm}^{-2}$	156 $\mu\text{J cm}^{-2}$
Cavity length L	100 μm	120 μm	40 μm	92 μm
n_g	4.6	3.2	7.8	3.9
Q factor	4250	2360	3700	3140

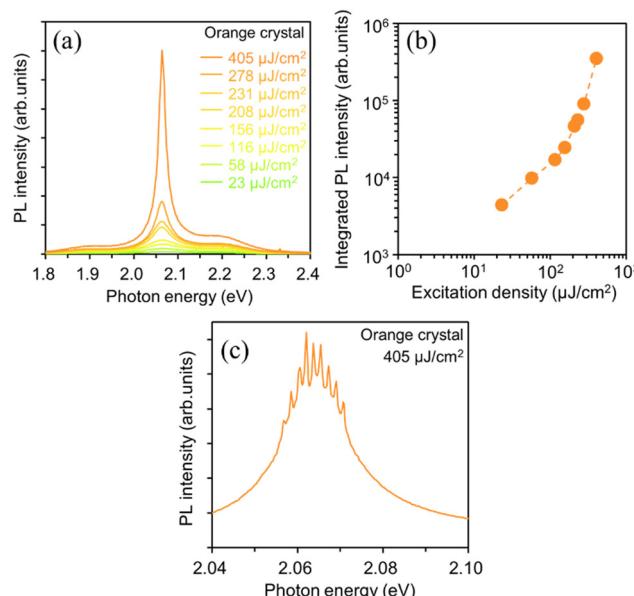


Fig. 10 Excitation density dependence of PL spectra (a) and integrated 0-2 PL band intensity (2.01–2.13 eV) (b) measured for the orange-emitting crystal. Lasing spectrum of the orange-emitting crystal measured at 405 $\mu\text{J cm}^{-2}$ (c).

density. Fig. 10(c) shows the high-resolution PL spectrum (spectral resolution: 0.2–0.3 nm) measured at 405 $\mu\text{J cm}^{-2}$ for the orange-emitting crystal. At 405 $\mu\text{J cm}^{-2}$, sharp lasing peaks appeared at the 0-2 emission band with a linewidth, $\Delta\nu$ of 14 cm^{-1} . The n_g value was calculated to be 3.9 from the $\Delta\nu$ value and the length of the circumference (92 μm) of the disk-shaped crystal. This n_g value is equivalent to the values obtained for the BP2T-CN crystals described above. This indicates that the observed lasing in the orange-emitting crystal is attributed to the whispering-gallery mode (WGM). The Q value was also estimated to be 3140 from the lasing spectrum, indicating that the orange-emitting crystal had a strong light confinement effect, similar to the other BP2T-CN crystals described above.

4. Conclusions

We prepared two types of BP2T-CN crystal polymorphs using the PVT and liquid-phase growth techniques. XRD measurements revealed that the green-emitting platelet-like and yellow-emitting

platelet-like crystals had monoclinic and triclinic forms, respectively. The changes in the molecular arrangements and intermolecular interactions resulted in the formation of green and yellow crystal polymorphs. By changing the crystal preparation method, the green platelet, yellow platelet, yellow fiber, and orange disk crystals were prepared. These four types of crystals displayed high PLQY values of 61–85%. In the green- and orange-emitting crystals that have the monoclinic form, the red-shifted emission in the orange-emitting crystal is attributed to the smaller slip angle. Multimode lasing was observed from the crystal end facets acting as F-P resonators in the green- and yellow-emitting (platelet, fiber) crystals. For the fiber-like crystal, lasing, at a lower threshold of 17 $\mu\text{J cm}^{-2}$ was obtained, owing to two-dimensional optical confinement effect and a higher absorption efficiency. In the fiber-like crystal, exciton polariton formation was also demonstrated by $E-k_z$ dispersion plots obtained from the energies of each interference peak in the mode structure of the F-P resonator. In the orange-emitting crystal, high Q -factor (3140) WGM lasing was observed above the threshold excitation density of 156 $\mu\text{J cm}^{-2}$. The results obtained herein show that the optical properties dependent on the crystal structure and molecular alignment can be obtained by controlling the molecular packing and magnitude of intermolecular interactions *via* a change in the crystal growth conditions.

Author contributions

H. Mizuno designed this study. H. Mizuno and T. Jinjyo constructed an optical setup for a laser-spectroscopy system. T. Jinjyo performed all experimental work and analyzed data with the aid of H. Mizuno. T. Jinjyo wrote the manuscript with contributions from H. Mizuno, F. Sasaki and H. Yanagi.

Conflicts of interest

There are no conflicts to declare.

Acknowledgements

This work was supported by JSPS KAKENHI Grant Number 21H01370 (Grant-in-Aid for Scientific Research(B)). This article is partially based on the results obtained from a project, JPNP20004, commissioned by the New Energy and Industrial



Technology Development Organization (NEDO). The work was also supported by JST CREST (JPMJCR02T4). The authors acknowledge Mr Y. Okajima, Nara Institute of Science and Technology, for the support in time-resolved PL measurements, Mr S. Katao, Nara Institute of Science and Technology, for XRD measurements, and Mr M. Fujihara, Nara Institute of Science and Technology, for the support of TG-DTA. We would like to thank Editage (www.editage.com) for the English language editing.

Notes and references

- 1 C. Kitamura, *Chem. Rec.*, 2012, **12**, 506–514.
- 2 G. Zhang, J. Lu, M. Sabat and C. L. Fraser, *J. Am. Chem. Soc.*, 2010, **132**, 2160–2162.
- 3 Z. Xie, H. Wang, F. Li, W. Xie, L. Liu, B. Yang, L. Ye and Y. Ma, *Cryst. Growth Des.*, 2007, **7**, 2512–2516.
- 4 K. Jo, S. Lee, A. Yi, T.-Y. Jeon, H. H. Lee, D. Moon, D. M. Lee, J. Bae, S.-T. Hong, J. Gene, S. G. Lee and H. J. Kim, *ACS Omega*, 2019, **4**, 19705–19709.
- 5 A. O. F. Jones, B. Chattopadhyay, Y. H. Geerts and R. Resel, *Adv. Funct. Mater.*, 2016, **26**, 2233–2255.
- 6 Q. Liao, X. G. Wang, S. Lv, Z. Xu, Y. Zhang and H. Fu, *ACS Nano*, 2018, **12**, 5359–5367.
- 7 C.-J. Ou, X.-H. Ding, Y.-X. Li, C. Zhu, M.-N. Yu, L.-H. Xie, J.-Y. Lin, C.-X. Xu and W. Huang, *J. Phys. Chem. C*, 2017, **121**, 14803–14810.
- 8 L. Pithan, C. Cocchi, H. Zschiesche, C. Weber, A. Zykov, S. Bommel, S. J. Leake, P. Schäfer, C. Draxl and S. Kowarik, *Cryst. Growth Des.*, 2015, **15**, 1319–1324.
- 9 Q. Qi, J. Zhang, B. Xu, B. Li, S. X.-A. Zhang and W. Tian, *J. Phys. Chem. C*, 2013, **117**, 24997–25003.
- 10 J. A. Schneider, H. Black, H. P. Lin and D. F. Perekhchka, *ChemPhysChem*, 2015, **16**, 1173–1178.
- 11 S. Shen, G. Xia, Z. Jiang, Q. Shao, W. Shan and H. Wang, *Cryst. Growth Des.*, 2018, **19**, 320–327.
- 12 S. Varghese, S. K. Park, S. Casado, R. Resel, R. Wannemacher, L. Lüer, S. Y. Park and J. Gierschner, *Adv. Funct. Mater.*, 2016, **26**, 2349–2356.
- 13 X. Wang, Z.-Z. Li, S.-F. Li, H. Li, J. Chen, Y. Wu and H. Fu, *Adv. Opt. Mater.*, 2017, **5**, 1700027.
- 14 Z. Zhang, X. Song, S. Wang, F. Li, H. Zhang, K. Ye and Y. Wang, *J. Phys. Chem. Lett.*, 2016, **7**, 1697–1702.
- 15 J. E. Park, M. Son, M. Hong, G. Lee and H. C. Choi, *Angew. Chem.*, 2012, **124**, 6489–6494.
- 16 W. Yao and Y. S. Zhao, *Nanoscale*, 2014, **6**, 3467–3473.
- 17 Y. S. Zhao, J. Xu, A. Peng, H. Fu, Y. Ma, L. Jiang and J. Yao, *Angew. Chem., Int. Ed.*, 2008, **47**, 7301–7305.
- 18 R. Ding, M. H. An, J. Feng and H. B. Sun, *Laser Photonics Rev.*, 2019, **13**, 1900009.
- 19 T. Hasegawa and J. Takeya, *Sci. Technol. Adv. Mater.*, 2009, **10**, 024314.
- 20 S. Mondal, W.-H. Lin, Y.-C. Chen, S.-H. Huang, R. Yang, B.-H. Chen, T.-F. Yang, S.-W. Mao and M.-Y. Kuo, *Org. Electron.*, 2015, **23**, 64–69.
- 21 K. Sugahara, T. Nakagawa, R. Hirase, T. Katagiri, Y. Inada, T. Yamao and S. Hotta, *Jpn. J. Appl. Phys.*, 2018, **57**, 04FL02.
- 22 M. A. Baldo, M. E. Thompson and S. R. Forrest, *Nature*, 2000, **403**, 750–753.
- 23 S. K. Park, J. H. Kim and S. Y. Park, *Adv. Mater.*, 2018, **30**, e1704759.
- 24 Q. Wei, N. Fei, A. Islam, T. Lei, L. Hong, R. Peng, X. Fan, L. Chen, P. Gao and Z. Ge, *Adv. Opt. Mater.*, 2018, **6**, 1800512.
- 25 B. Geffroy, P. le Roy and C. Prat, *Polym. Int.*, 2006, **55**, 572–582.
- 26 Y. Cui, H. Yao, J. Zhang, K. Xian, T. Zhang, L. Hong, Y. Wang, Y. Xu, K. Ma, C. An, C. He, Z. Wei, F. Gao and J. Hou, *Adv. Mater.*, 2020, **32**, 1908205.
- 27 L. Hong, H. Yao, Z. Wu, Y. Cui, T. Zhang, Y. Xu, R. Yu, Q. Liao, B. Gao, K. Xian, H. Y. Woo, Z. Ge and J. Hou, *Adv. Mater.*, 2019, **31**, e1903441.
- 28 M. Li, A. H. Balawi, P. J. Leenaers, L. Ning, G. H. L. Heintges, T. Marszalek, W. Pisula, M. M. Wienk, S. C. J. Meskers, Y. Yi, F. Laquai and R. A. J. Janssen, *Nat. Commun.*, 2019, **10**, 2867.
- 29 Z. Chen, C. Dai, W. Xiong, Y. Che and C. Zhang, *Commun. Chem.*, 2021, **4**, 97.
- 30 M. Chu, B. Qiu, W. Zhang, Z. Zhou, X. Yang, Y. Yan, J. Yao, Y. J. Li and Y. S. Zhao, *ACS Appl. Mater. Interfaces*, 2018, **10**, 42740–42746.
- 31 H. Liu, Z. Bian, Q. Cheng, L. Lan, Y. Wang and H. Zhang, *Chem. Sci.*, 2019, **10**, 227–232.
- 32 J.-J. Wu, X.-D. Wang and L.-S. Liao, *ACS Photonics*, 2019, **6**, 2590–2599.
- 33 S. Zhao, H. Yamagishi, O. Oki, Y. Ihara, N. Ichiji, A. Kubo, S. Hayashi and Y. Yamamoto, *Adv. Opt. Mater.*, 2021, **10**, 2101808.
- 34 H. Mizuno, I. Ohnishi, H. Yanagi, F. Sasaki and S. Hotta, *Adv. Mater.*, 2012, **24**, 2404–2408.
- 35 H. Mizuno, U. Haku, Y. Marutani, A. Ishizumi, H. Yanagi, F. Sasaki and S. Hotta, *Adv. Mater.*, 2012, **24**, 5744–5749.
- 36 Y. Tanaka, K. Goto, K. Yamashita, T. Yamao, S. Hotta, F. Sasaki and H. Yanagi, *Appl. Phys. Lett.*, 2015, **107**, 163303.
- 37 R. Hatano, K. Goto, K. Yamashita, F. Sasaki and H. Yanagi, *Jpn. J. Appl. Phys.*, 2017, **56**, 04CL02.
- 38 S. Dokiya, F. Sasaki and H. Yanagi, *J. Cryst. Growth*, 2017, **468**, 792–795.
- 39 H. Mizuno, T. Jinjyo, C. M. Laurio, H. Katsuki, I. Hiromitsu, F. Sasaki and H. Yanagi, *Jpn. J. Appl. Phys.*, 2020, **59**, SDDA14.
- 40 V. A. Postnikov, Y. I. Odarchenko, A. V. Iovlev, V. V. Bruevich, A. Y. Pereverzev, L. G. Kudryashova, V. V. Sobornov, L. Vidal, D. Chernyshov, Y. N. Luponosov, O. V. Borshchev, N. M. Surin, S. A. Ponomarenko, D. A. Ivanov and D. Y. Paraschuk, *Cryst. Growth Des.*, 2014, **14**, 1726–1737.
- 41 T. Katagiri, S. Ota, T. Ohira, T. Yamao and S. Hotta, *J. Heterocycl. Chem.*, 2007, **40**, 853–862.
- 42 S. Hotta, M. Goto, R. Azumi, M. Inoue, M. Ichikawa and Y. Taniguchi, *Chem. Mater.*, 2004, **16**, 237–241.
- 43 S. Kanazawa, M. Ichikawa, T. Koyama and Y. Taniguchi, *ChemPhysChem*, 2006, **7**, 1881–1884.



44 S. Dokiya, F. Sasaki, S. Hotta and H. Yanagi, *Jpn. J. Appl. Phys.*, 2016, **55**, 03DC13.

45 S. Dokiya, H. Ishigami, T. Akazawa, F. Sasaki and H. Yanagi, *Jpn. J. Appl. Phys.*, 2020, **59**, 041004.

46 C.-F. Liu, X. Liu, W.-Y. Lai and W. Huang, *Adv. Mater.*, 2018, **30**, 1802466.

47 Q. Ou, Q. Peng and Z. Shuai, *Nat. Commun.*, 2020, **11**, 4485.

48 H. Mizuno, T. Jinjyo, K. Bando, F. Sasaki, K. Yamashita and H. Yanagi, *J. Mater. Chem. C*, 2021, **9**, 11189–11197.

49 M. Kasha, Molecular Excitons in Small Aggregates, in *Spectroscopy of the Excited State*, ed. B. Di Bartolo, D. Pacheco and V. Goldberg, 1976, pp. 337–363.

50 K. Bando, T. Nakamura, S. Fujiwara, Y. Masumoto, F. Sasaki, S. Kobayashi, Y. Shimoi and S. Hotta, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2008, **77**, 045205.

51 K. Takazawa, J. Inoue, K. Mitsuishi and T. Takamasu, *Phys. Rev. Lett.*, 2010, **105**, 067401.

52 Q. Liao, Z. Xu, X. Zhong, W. Dang, Q. Shi, C. Zhang, Y. Weng, Z. Li and H. Fu, *J. Mater. Chem. C*, 2014, **2**, 2773–2778.

53 X. Wang, Q. Liao, Z. Xu, Y. Wu, L. Wei, X. Lu and H. Fu, *ACS Photonics*, 2014, **1**, 413–420.

54 Q. Shang, S. Zhang, Z. Liu, J. Chen, P. Yang, C. Li, W. Li, Y. Zhang, Q. Xiong, X. Liu and Q. Zhang, *Nano Lett.*, 2018, **18**, 3335–3343.

