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## Influence of surfactant on glass transition temperature of poly(lactic-co-glycolic acid) nanoparticles†

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Poly(*D,L*-lactic-co-glycolic acid) (PLGA) is one of the most commonly used drug carriers in nanomedicines because of its biodegradability, biocompatibility and low toxicity. However, the physico-chemical characterization and study of drug release are often lacking the investigation of the glass transition temperature ( $T_g$ ), which is an excellent indicator of drug release behavior. In addition, the residual surfactant used during the synthesis of nanoparticles will change the glass transition temperature. We thus prepared PLGA nanoparticles with polymeric (poly(vinyl alcohol) (PVA)) and ionic (didodecyldimethylammonium bromide (DMAB)) surfactant to investigate their influence on the glass transition temperature. Determination of  $T_g$  in dry and wet conditions were carried out. The use of concentrated surfactant during synthesis resulted in a larger amount of residual surfactant in the resulting particles. Increasing residual PVA content resulted in an increase in particle  $T_g$  for all but the most concentrated PVA concentrations, while increasing residual DMAB content resulted in no significant change in particle  $T_g$ . With the presence of residual surfactant, the  $T_g$  of particle and bulk samples measured in wet conditions is much lower than that in dry conditions, except for bulk PLGA containing the ionic surfactant, which may be related to the plasticizing effect of the DMAB molecules. Notably, the  $T_g$  of both particles in wet conditions is approaching physiological temperatures where subtle changes in  $T_g$  could have dramatic effects on drug release properties. In conclusion, the selection of surfactant and the remaining amount of surfactant are crucial parameters to utilize in designing the physico-chemical properties of PLGA particles.

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## 1. Introduction

Polymeric particles are adaptable vehicles for drug delivery since the drug release kinetics can be purposely designed by modifying chemical properties, such as monomer composition and molecular weight, as well as physical properties, such as particle size, morphology, and porosity of the formulation. Among polymeric particles, poly(lactic-co-glycolic acid) (PLGA) has proven to be a very successful drug delivery system in the past few decades.<sup>1–4</sup> PLGA is a random copolymer consisting of a defined ratio of poly(lactic acid) and poly(glycolic acid) that has been applied extensively in biomedical products due to its biocompatibility, biosafety, and degradability. Many PLGA formulations, from millimeter to nanometer size, are available by different manufacturing processes,<sup>5–9</sup> and PLGA particles can encapsulate a wide range of drug molecules.

PLGA particles are usually characterized by particle size, size distribution, zeta potential, morphology, and drug loading efficiency, as these are crucial factors that contribute to the drug release profiles.<sup>10</sup> Glass transition temperature ( $T_g$ ), however, is an underutilized property in PLGA particle characterization.<sup>11</sup> When the surrounding temperature of PLGA is higher than the glass transition temperature, more drug is rapidly released shortly after incubation as a burst release of drug. This increase in burst release is because the PLGA transforms from a glassy solid into a soft rubbery material while above the  $T_g$ , allowing for more drug loaded inside the PLGA to be suddenly released due to the enhanced polymer chain mobility. The long-term drug release is based on PLGA's biodegradation. The kinetics of drug release associated with biodegradation is affected by many factors, such as the molecular weight of the copolymer,<sup>12</sup> monomer ratio, hydrophobicity of drug, manufacturing process, and release medium.<sup>5,6,13</sup> It is also indicated that the closer the  $T_g$  is to body temperature, the worse long-term drug release will be.<sup>14</sup>

The analysis of  $T_g$  in PLGA particles is complicated by the presence of a surfactant. PLGA particle preparation techniques, such as nanoemulsion, require the use of a surfactant to stabilize the system and prevent the aggregation of the particles. Polyvinyl

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alcohol (PVA) is the most commonly used surfactant in PLGA particle formulation. However, previous studies have shown that a portion of PVA stays with PLGA nanoparticles even after washing.<sup>15,16</sup> Notably, residual PVA can change the particle  $T_g$  by introducing a second polymeric component to the system. As an alternative surfactant, didodecyldimethylammonium bromide (DMAB) is also widely implemented to generate PLGA particles.<sup>17–20</sup> DMAB is a cationic surfactant that will impart a positive charge on the particle surface, which has been proposed to enhance interactions between cells and nanoparticles. For instance, Jin *et al.* made drug loaded 100 nm PLGA nanoparticles with DMAB and achieved a significant increase in cellular uptake due to the positively charged surface.<sup>19</sup> However, residual surfactant on the surface of a NP can alter the interfacial interactions and mobility of the polymer chain, thus affecting  $T_g$  and therefore drug release properties.

Another complication in analyzing polymeric particles'  $T_g$  is the addition of small molecules which can act as plasticizers.<sup>21,22</sup> The mechanical properties of polymers can be changed through the addition of small compounds that can embed themselves into the polymer matrix and distribute throughout it. Plasticizers can lower the intermolecular interactions of polymers, resulting in an increase in mobility of polymer chains, and as a result, a decrease in  $T_g$ .<sup>23,24</sup> When PLGA nanoparticles are administered as drug delivery vehicles in biological fluids, the presence of water can have a significant impact on the  $T_g$  of PLGA, as water can act as a plasticizer and increase the mobility of the polymer chains. Therefore, it is important to measure  $T_g$  of PLGA nanoparticles in wet conditions to gain a better understanding of their behavior during drug delivery and to optimize their design for biomedical applications.

In the present study, we prepared PLGA nanoparticles with PVA (PVA-PLGA NPs) and DMAB (DMAB-PLGA NPs) and determined the amount of residual surfactant present. Additionally, the  $T_g$  of PVA-PLGA NPs and DMAB-PLGA NPs in both dry and wet conditions were determined. Furthermore, the  $T_g$  of the bulk was compared to the  $T_g$  of the nanoparticles to illustrate the deviations due to 3D confinement and the importance of measuring  $T_g$  for designing drug delivery nanoparticles.

## 2. Methodology

### 2.1 Materials

The PLGA (85:15, molecular weight: 50 000–75 000 Da), PVA (molecular weight: 32 000–50 000, 87–89% hydrolyzed), and chloroform (HPLC grade, amylene stabilized) were purchased from Sigma-Aldrich. DMAB (99% purity), DMSO-d6 (methyl sulfoxide, 99.9% atom D), and chloroform-d (99.8% atom D, 0.03 v/v% TMS) were purchased from Thermo Scientific. Ultrapure water (Thermo Scientific Smart2Pure 3 UV/UF) was applied in particle preparation and purification.

### 2.2 Method

**2.2.1 Preparation of nanoparticles.** PVA-PLGA NPs were prepared by a single emulsion technique. In brief, 20 mg of PLGA was dissolved in 1 ml of chloroform to form the initial

organic phase. The aqueous phase contained 0.5 w/v% to 2.5 w/v% PVA to investigate the influence of surfactant on the particle properties. The organic phase was injected into 4 ml of the aqueous phase and the mixture was sonicated (Q700, Qsonica) under 100 amplitude for 20 minutes (1 second on and 4 seconds off, 100 minutes of total run time). The obtained emulsion was stirred overnight to evaporate the organic solvent.

DMAB-PLGA NPs was synthesized by modifying the method presented by Kwon. *et al.*<sup>17</sup> In detail, 20 mg of PLGA was dissolved in 1 ml chloroform. Then, the organic phase was added into 4 ml of aqueous solution with varying amounts of DMAB (0.25–2 w/v%). The mixture was emulsified by tip sonication under 100 amplitude for 20 minutes (1 second on and 4 second off, 100 minutes of total run time). The obtained emulsion was poured into 8 ml of ultrapure water and stirred overnight. The particles were washed by five centrifugation cycles using an Eppendorf Centrifuge 5430 R at 22 136 RCF for one hour. After each centrifugation cycle, the supernatant was removed and the particles were resuspended in pure water.

For measuring the  $T_g$  of non-PVA PLGA nanoparticles, nanoprecipitation was used to produce surfactant free nanoparticles. Briefly, 20 mg of PLGA was measured and dissolved in 2 ml of acetone to form the organic phase. Subsequently, the organic phase was introduced into 100 ml of ultrapure water with stirring. The resulting mixture was stirred overnight to allow complete evaporation of the organic solvent. Finally, the samples were filtered using a 40  $\mu$ m filter to eliminate any large debris.

**2.2.2 Size and morphology study.** Nanoparticle tracking analysis (NTA, NS300) was applied to measure the nanoparticle size and size distribution. In brief, 2 ml of the particle suspension was collected after purification and diluted between 20 to 200 times to ensure the concentration was appropriate for detection (approximately 10<sup>8</sup> particles per ml). For each particle sample, five 1 minute videos were recorded to capture the movement of the particles.

Scanning electron microscopy (SEM, JSM-7900F) was used to observe the morphology of PVA-PLGA and DMAB-PLGA NPs. 10  $\mu$ l of particle suspension was deposited on a small piece of aluminium foil and dried in the fume hood. Then, the sample was coated with carbon to generate a conductive surface and imaged by SEM.

**2.2.3  $T_g$  measurement by modulated differential scanning calorimetry (mDSC).** Modulated DSC (Discovery DSC 250, TA instrument) is an extension of traditional DSC which introduces a sinusoidal modulation to the linear ramp.<sup>25</sup> The temperature modulation varies the heating process in a periodic manner which provides signals on reversing and non-reversing heat flow. mDSC allows for the separation of any aging peaks that overlap with the  $T_g$ , as a common issue with  $T_g$  analysis using conventional DSC is the overlapping of the glass transition and molecular relaxation peak.<sup>26</sup>  $T_g$  was determined by analyzing the reversing heat flow. PLGA samples (~3 mg) were sealed inside a Tzero pan with a lid. An empty reference pan was prepared and sealed as well. For  $T_g$  measurements in wet conditions, 30  $\mu$ l of ultrapure water was added to the Tzero hermitic pan to create an aqueous suspension. The average heating rate was 1 °C min<sup>-1</sup> and the amplitude was 1 °C. All measurements were operated under

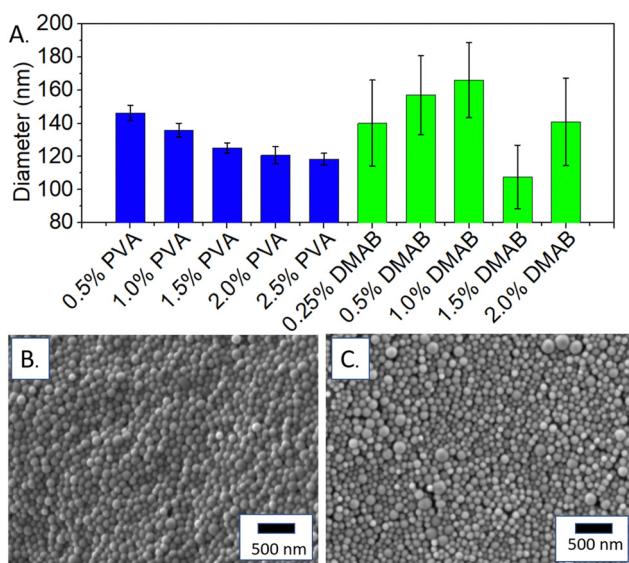
pure nitrogen.  $T_g$  was determined by analyzing the inflection point from the reversing heat flow curve.

**2.2.4  $^1\text{H}$  nuclear magnetic resonance spectroscopy (NMR).** Residual surfactant quantification by  $^1\text{H}$  NMR was performed using a Bruker AVIII-500. PVA-PLGA samples were weighed ( $\sim 3$  mg) and dissolved in DMSO-d6 (0.6 ml) while DMAB-PLGA samples were dissolved in deuterated chloroform (0.6 ml).

### 3. Results and discussion

#### 3.1 Characterization of PLGA nanoparticle diameter

PLGA nanoparticles were prepared with different polymeric (PVA) and cationic (DMAB) surfactant concentration. The mean size as a function of the surfactant concentration used during synthesis is displayed in Fig. 1A. Representative SEM images of particles prepared by PVA (Fig. 1B) and DMAB (Fig. 1C) confirmed that particles were successfully synthesized. For PVA-PLGA NPs, PVA concentrations ( $C_{\text{PVA}}$ ) of 0.5%, 1%, 1.5%, 2% and 2.5% in the aqueous phase were used during nanoemulsion preparation of the particles. PVA plays an important role in nanoparticle preparation by preventing aggregation of the PLGA.<sup>27,28</sup> The particle diameter shows a steady decrease as higher PVA concentration was used during particle synthesis, with mean diameters of  $146.4 \pm 5.1$  nm,  $136.0 \pm 4.7$  nm,  $125.1 \pm 3.3$  nm, and  $120.7 \pm 6.0$  nm, with increasing  $C_{\text{PVA}}$  from 0.5 v/w% to 2.0 v/w%. The particle diameter with 2.5 w/v% initial PVA of  $118.4 \pm 4.0$  nm is statistically similar to that of 2.0 w/v% initial PVA (Fig. S2, ESI†). Miladi *et al.* has reported a decrease in the mean diameter of nanoparticles with increasing PVA concentrations, explained by increasing viscosity of the aqueous phase with increasing PVA concentration, preventing



**Fig. 1** Size of PLGA nanoparticles made using PVA (blue) and DMAB (green) (mean  $\pm$  SD,  $n = 3$ ). Size decreases with increasing PVA concentration while particles prepared with DMAB have no clear trend and large error bars, indicating that the size of DMAB-PLGA particles is more variable. SEM image of PVA-PLGA NPs (B) and SEM image of DMAB-PLGA NPs (C).

coalescence of emulsion droplets formed by emulsification.<sup>29</sup> Therefore, a rise in the concentration of PVA ensures a greater stability of the system, and as a result, a reduction in the coalescence of the emulsion.<sup>28</sup> It has also been suggested that when increasing the surfactant concentration, a relatively small amount of stabilizer is adsorbed at the interface, while the surplus remains within the aqueous phase without exerting any notable influence on the emulsification process,<sup>17,30</sup> which explains the similar particle diameters found with 2.0 w/v% and 2.5 w/v% initial PVA concentration. Moreover, Sahoo *et al.* discovered a positive correlation between initial PVA concentration used during nanoemulsion particle synthesis and residual surfactant present on the resulting nanoparticles; even after purification, a certain amount of residual PVA was associated with the nanoparticles. As a result, increased residual PVA will stabilize the emulsion droplets, leading to smaller nanoparticles.<sup>15</sup> For DMAB-PLGA nanoparticles, the diameter fluctuates as the initial DMAB concentration increases (Fig. 1A). The error bars from each data point suggest that the particle diameter of DMAB-PLGA particles is not as stable as PVA-PLGA particles. It has been reported that the utilization of 0.5% DMAB in ethyl acetate during the nanoemulsion process resulted in the production of particles of approximately 50 nm. Subsequently, an increase in DMAB concentration to 1% led to a corresponding increase in particle size of 102 nm, suggesting that the cationic nature of DMAB favors its retention in the aqueous phase rather than participating in the emulsification process, resulting in DMAB playing a less significant role in emulsification.<sup>31</sup> In Cooper *et al.*'s study, DMAB-stabilized PLGA nanoparticles were synthesized with concentrations ranging from 0.1% to 1%. The investigation demonstrated that the nanoparticle size was maximized when utilizing DMAB concentrations between 0.25% and 0.75% w/v, with particle aggregation observed at DMAB concentrations exceeding 1%.<sup>32</sup> However, Kwon *et al.* reported that DMAB-PLGA particle diameter decreases substantially from 1 w/v% to 2 w/v% initial DMAB concentration.<sup>17</sup> The results presented here also support a significant decrease in diameter occurring at higher initial DMAB concentration of 1.5 w/v%. However, the trend does not continue at 2 w/v% initial DMAB concentration.

#### 3.2 Evaluation of residual surfactant

A significant amount of residual surfactant is known to remain on the PLGA particle surface or within the particle,<sup>15</sup> despite washing, and this has been shown to alter particle attributes such as intracellular uptake and hydrophobicity. Typical washing procedures of PLGA NPs usually consist of 3 washing cycles.<sup>33</sup> However, our  $T_g$  results of 3-times washed particles showed a second  $T_g$  around 68 °C (Fig. 2), corresponding to the  $T_g$  of PVA, which indicated that there was still a significant amount of non-blended PVA in PLGA NPs. Thus, increasing the washing cycles was needed to remove free PVA from PLGA nanoparticles so only one  $T_g$  was observed.

$^1\text{H}$  NMR was used to determine the amount of residual surfactant. The  $^1\text{H}$  NMR spectrum of 2% PVA-PLGA is shown in Fig. 3, where peak A at 5.2 ppm is related to the CH of the

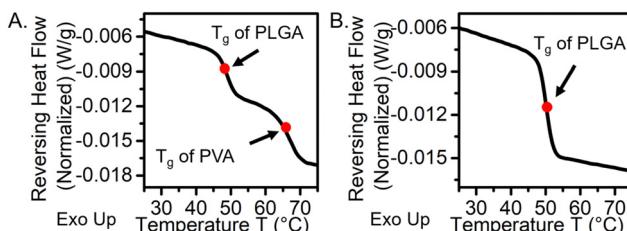


Fig. 2 mDSC thermograph of PVA-PLGA NPs washed 3 times (A) and 5 times (B). A second transition was observed from 3-times washed PLGA particles (A). Unblended PVA was removed after 5 washes (B).

lactide unit while peak B around 4.8 ppm corresponds to the  $\text{CH}_2$  of the glycolide unit.<sup>34,35</sup> The peak (C) around 3.8 ppm is assigned to the CH from PVA. The peaks at 3.3 ppm and 2.5 ppm are assigned to  $\text{H}_2\text{O}$  and DMSO, respectively. Additionally,  $^1\text{H}$  NMR was also performed on DMAB-PLGA nanoparticles. The spectrum of 2% DMAB-PLGA is presented in Fig. 3, where the peak (D) at 3.4 ppm is assigned to the two methyl groups in the DMAB molecule. The sharp peak at 7.3 ppm is related to chloroform. The following equations were performed to calculate residual surfactant:

$$\text{PVA\%} = \frac{P_{\text{PVA}} \cdot M_{\text{PVA}}}{P_{\text{PVA}} \cdot M_{\text{PVA}} + P_{\text{PLA}} \cdot M_{\text{PLA}} + \frac{P_{\text{PGA}} \cdot M_{\text{PGA}}}{2}} \quad (1)$$

$$\text{DMAB\%} = \frac{P_{\text{PVA}} \cdot M_{\text{PVA}}}{\frac{P_{\text{DMAB}} \cdot M_{\text{DMAB}}}{6} + P_{\text{PLA}} \cdot M_{\text{PLA}} + \frac{P_{\text{PGA}} \cdot M_{\text{PGA}}}{2}} \quad (2)$$

where  $P_{\text{PVA}}$ ,  $P_{\text{PLA}}$ ,  $P_{\text{PGA}}$ , and  $P_{\text{DMAB}}$  are the peak integrations of the selected peak from each component.  $M_{\text{PVA}}$ ,  $M_{\text{PLA}}$ , and  $M_{\text{PGA}}$  are the molecular weights of the monomers of PVA, PLA, and PGA, respectively.  $M_{\text{DMAB}}$  is the molecular weight of DMAB. In Fig. 4C, residual DMAB showed an increasing trend from 4.6% to 8.6% with increasing initial DMAB concentration. Residual PVA varied from 5.4% to 8.7% when initial PVA concentration changed from 0.5% to 2.5% (Fig. 4D).

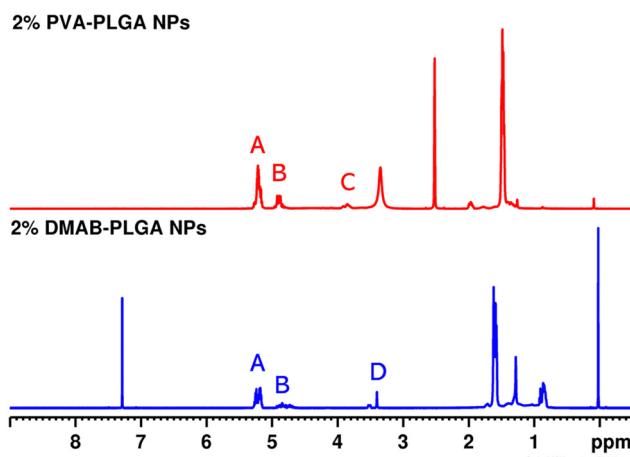


Fig. 3 NMR spectra of 2% PVA-PLGA NPs in  $\text{DMSO-d}_6$  (red) and 2% DMAB-PLGA NPs in  $\text{CDCl}_3$  (blue).

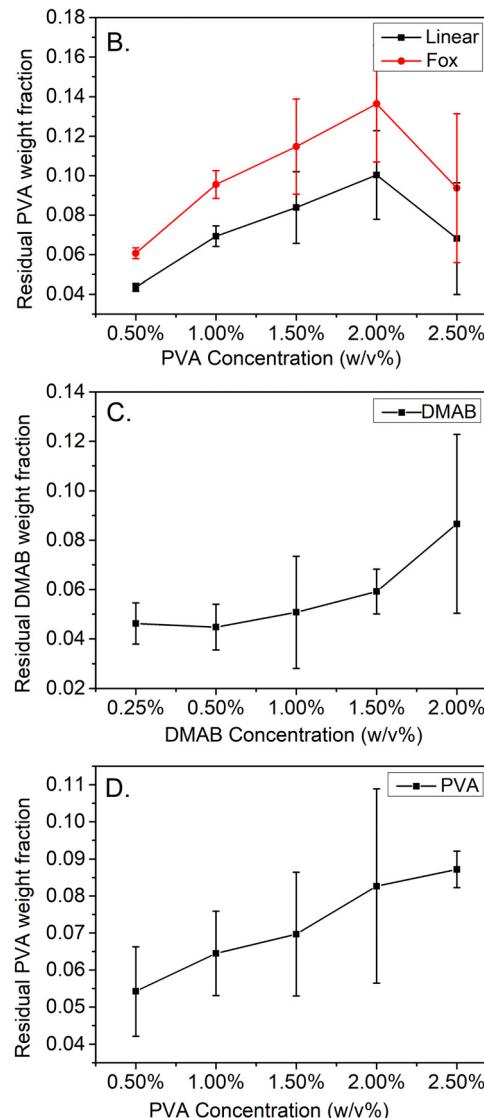
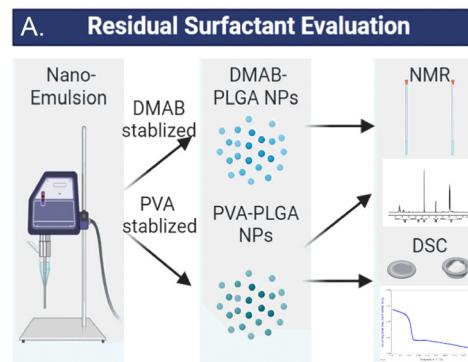


Fig. 4 Residual surfactant evaluation by mDSC and NMR (A). Residual PVA was calculated by the linear equation and Fox equation (B) which determined the lower and upper limits of weight percentage. NMR was also used to determine residual DMAB (C) and residual PVA (D) (mean  $\pm$  SD,  $n = 3$ ).

Despite the presence of one  $T_g$  in the DSC scan, residual PVA still remains but has blended with the PLGA, increasing the  $T_g$ . Thus, mDSC can also be used to estimate the quantity of

residual PVA that has blended with the PLGA in the particles (Fig. 4A). Using mDSC to determine the  $T_g$  of PVA bulk, PVA-PLGA nanoemulsion particles, and non-surfactant PLGA particles, the amount of residual surfactant can be estimated. For this analysis, only the third heating curves of the mDSC scans were analyzed, which was confirmed to be only from PLGA bulk (Fig. 5), thus the influences from the confinement as a particle, interfacial effects, and the thermal history on the  $T_g$  were able to be eliminated. The bulk  $T_g$  of PVA-PLGA particles is greater than the bulk  $T_g$  for surfactant-free PLGA particles, indicating that the PVA ( $T_g$ : 68.8 °C) had blended with the PLGA. Accordingly, the following formulas were used to calculate the predicted weight % of residual PVA:

Linear equation:

$$T_{g,\text{blend}} = w_1 \cdot T_1 + w_2 \cdot T_2 \quad (3)$$

Fox equation:<sup>36</sup>

$$\frac{1}{T_{g,\text{blend}}} = \frac{w_1}{T_1} + \frac{w_2}{T_2} \quad (4)$$

where  $T_{g,\text{blend}}$  is the bulk  $T_g$  of the PLGA particles,  $w_1$  is the weight fraction of PLGA in the particles,  $T_1$  is the bulk  $T_g$  of non-surfactant PLGA particles ( $T_g = 48.3$  °C,  $n = 3$ ),  $w_2$  is the weight fraction of PVA in the particles, and  $T_2$  is the  $T_g$  of PVA bulk. The results show that the residual PVA varies from 4% to 10% calculated by the linear equation with the initial PVA concentrations from 0.2% to 2.5% (Fig. 4B), while the Fox equation results in a slightly higher percentage of residual PVA of 6% to 13.6% (Fig. 4B). The DSC evaluation implies that the amount of residual PVA dropped at 2.5% initial PVA concentration. This decrease in residual surfactant observed by  $T_g$  analysis is contrary to the NMR results which show an increase in residual surfactant. This apparent contradiction could

be explained by phase separation as the binary polymer blend begins to phase separate with increasing PVA concentration,<sup>37</sup> which gives a relatively lower  $T_g$  as not all the PVA is blended with the PLGA. By comparing the DSC estimates of the weight percentage of surfactant to the values obtained from NMR analysis, the degree of blending can be evaluated. Specifically, if the weight fraction from DSC equals the value obtained from NMR, it can be concluded that all PVA is blended. Otherwise, a smaller weight fraction from DSC indicates that a certain proportion of PVA is not blended. Thus,  $T_g$  analysis by DSC can provide an estimate of whether the PVA is blending with the PLGA or not.

### 3.3 Determination of mDSC annealing temperature

Modulated DSC (mDSC) was used to determine the nanoparticle  $T_g$  and the bulk  $T_g$  of the sample. The  $\Delta T_g$  is calculated as the nanoparticle  $T_g$  - bulk  $T_g$ . A pre-treatment step was applied for  $T_g$  measurement of PLGA nanoparticles and bulk before the heating ramp to erase the thermal history. The pre-treatment step is similar to that which Zhang *et al.* and Christie *et al.* implemented in order to determine the  $T_g$  of PS particles; the samples are subjected to a pre-treatment step at a temperature that is relatively close to the  $T_g$  in order to erase the thermal history of the samples, but the samples will continue to exist in the form of particles.<sup>38,39</sup> In this study, a pre-treatment temperature of 48 °C is used for erasing the thermal history of the particles in dry conditions and 38 °C for particles in wet conditions. The morphology of particles after pre-treatment, shown in the SEM inserts in Fig. 5, confirms that particles still remain after the pre-treatment step. The pre-treatment process is included in the mDSC run as the first heating ramp. The nanoparticle  $T_g$  with thermal history erased is determined by using the second heating ramp. During the second heating, the sample is heated to 190 °C in dry conditions to complete the transformation of the sample into bulk PLGA. Even though crystallization is not anticipated in the PLGA used in this study, the second heating cycle brings the sample above the  $T_m$  of PLGA in order to guarantee that the bulk state is attained. To determine the bulk PLGA  $T_g$ , the third heating ramp is brought up to 80 °C for a clear analysis of the  $T_g$ . Similarly, for wet measurements the  $T_g$  of the nanoparticles is obtained from the second heating cycle of the mDSC measurements. To obtain the  $T_g$  of wet bulk, samples were freeze-dried in the pan to remove water and then heated to 190 °C to completely transform the particles to the bulk stage. Lastly, the post-treated sample was measured in wet conditions to obtain the corresponding  $T_g$  of the bulk sample. From intrinsic viscosity measurements it was determined that the DSC procedure did not result in significant degradation (Fig. S1, ESI†).

### 3.4 $T_g$ of PVA-PLGA and DMAB-PLGA nanoparticles

Fig. 6 shows the  $T_g$  of PVA-PLGA NPs and DMAB-PLGA NPs in dry conditions with varying weight fraction of residual surfactant (as determined by NMR) and the  $T_g$  with the corresponding particle diameters. In Fig. 6A,  $T_g$  increases with increasing residual PVA, except the  $T_g$  from particles synthesized by 2.5 w/v% PVA in the final data point with a weight fraction of

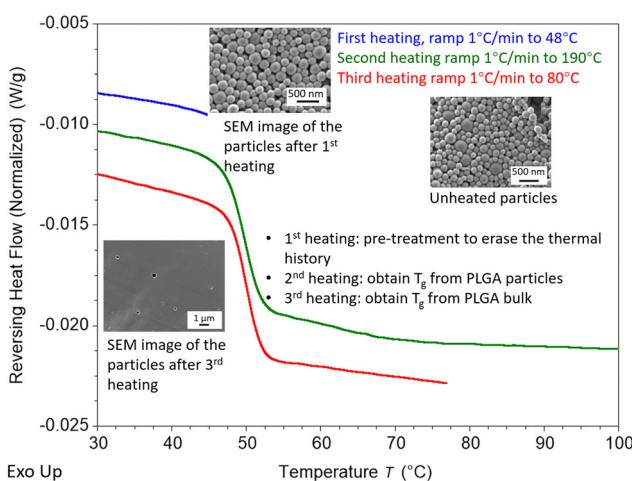


Fig. 5 DSC procedure and SEM images of PLGA nanoparticles. The blue curve represents the pre-treatment to erase thermal history while keeping the morphology of particles. Particle  $T_g$  was obtained from the second heating cycle (green curve). Bulk  $T_g$  was obtained from the third heating cycle (red curve). The SEM image of the sample after the third heating confirmed no particles existed in the last heating cycle. Curves shifted vertically for easier viewing.

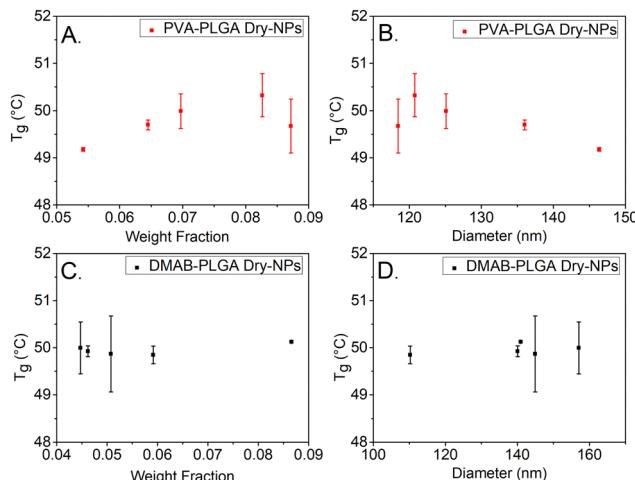


Fig. 6  $T_g$  of dry particles vs. residual surfactant (A and C) and  $T_g$  of dry particles vs. particle diameter (B and D) (mean  $\pm$  SD,  $n = 3$ ).

0.087 residual PVA. PVA-PLGA NPs prepared with concentrated surfactant have more residual surfactant, which results in an increased overall  $T_g$  of the binary system since the  $T_g$  of PVA is higher than PLGA's (Fig. S6, ESI<sup>†</sup>). However, phase separation can occur if the second polymeric component reaches its solubility limit, which suggests that not all the PVA is blended with the PLGA for the last data point. Fig. 6B indicates an opposite trend in  $T_g$  vs. weight fraction of residual surfactant because the mean diameter of the NPs decreases with increasing PVA concentration. The statistical analysis revealed a significant difference between the initial and final data points. The error bars, which represent the standard deviation from a sample size of  $n = 3$ , indicate consistent  $T_g$  results, despite the small magnitude of difference of approximately one degree between groups. The mDSC technique has been shown to yield more accurate results than standard DSC due to the removal of thermal relaxation which would result in  $T_g$  shift.<sup>26</sup> Fig. 6C shows no clear trend between the  $T_g$  of DMAB-PLGA NPs in dry conditions and residual DMAB, as one-way ANOVA and Tukey's post-test show no statistically significant difference among this data (Fig. S5, ESI<sup>†</sup>). Similarly, no significant difference is observed in Fig. 6D with particle diameter ranging from 110 nm to 158 nm.

As shown in Fig. 7A, the  $T_g$  of PVA-PLGA bulk in dry conditions experiences a subtle increase from 0.5% to 2.0% initial PVA concentration and a decrease from 2.0% to 2.5% initial PVA concentration compared to the  $T_g$  of NPs, but a trend still exists. The  $\Delta T_g$  of PVA-PLGA NPs in dry conditions gives a similar trend. However, the  $T_g$  of DMAB-PLGA bulk experiences a significant reduction (Fig. 7C) after the second heating cycle, which could be explained by the heating process promoting the penetration of DMAB into the PLGA, plasticizing the system. The decreasing trend is caused by the larger amount of residual DMAB in the particles. The positive correlation of  $\Delta T_g$  and DMAB concentration results from the opposite trends present in the  $T_g$  of DMAB-PLGA dry bulk and DMAB-PLGA dry nanoparticles. The  $\Delta T_g$  of PVA-PLGA samples (Fig. 7B) would suggest no clear confinement effect on  $T_g$ , which in

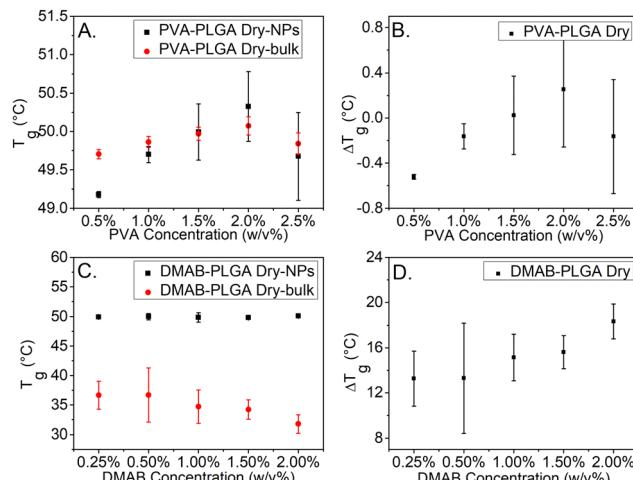


Fig. 7 Particle  $T_g$  and bulk  $T_g$  measured in dry condition for both PVA-PLGA (A) and DMAB-PLGA (C).  $\Delta T_g$  of PVA-PLGA (B) and DMAB-PLGA (D) in dry conditions (mean  $\pm$  SD,  $n = 3$ ).

contrast to the  $T_g$  results for polystyrene nanoparticles.<sup>38–40</sup> However, the size range of PLGA nanoparticles in this study is relatively narrow, within 50 nm, which limits the ability to observe a significant size effect. Additionally, the presence of residual surfactant affecting the  $T_g$  may be masking any confinement effects. Specifically, the  $\Delta T_g$  between PVA-PLGA and DMAB-PLGA is markedly different in the presence of surfactant.

Water is well-known to plasticize many polymers, including PLGA, and will therefore have a significant impact on  $T_g$  when present.<sup>12,41</sup> Similarly, the results presented here with PLGA particles show that the  $T_g$  of nanoparticles synthesized with PVA (Fig. 8A) and DMAB (Fig. 8C) is much lower when evaluated in water compared to dry conditions, suggesting that these materials will undergo plasticization when exposed to fluids during use as a drug delivery vehicle. Zhang *et al.* reported that polystyrene nanoparticles exhibit a  $T_g$  reduction in wet

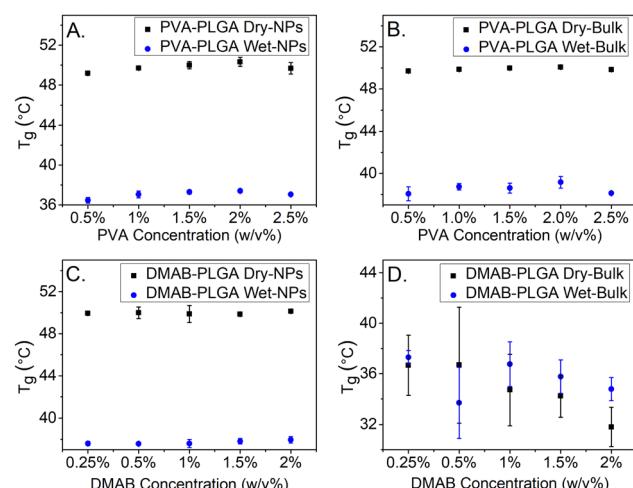


Fig. 8  $T_g$  of particles in dry and wet conditions vs. surfactant concentrations (A and C).  $T_g$  of bulk in dry and wet condition vs. surfactant concentrations (B and D) (mean  $\pm$  SD,  $n = 3$ ).

measurement conditions, which was caused by a mobile layer on the particle surface ( $\sim 30$  nm thickness).<sup>38</sup> It is likely that a similar phenomenon may be occurring here where the presence of water penetrates the surface of PLGA nanoparticles freeing up the polymer chains and increasing the chain mobility. They also suggested that significant  $T_g$  reduction was observed when the particle diameter was less than 600 nm. Given that wet conditions are crucial for drug delivery particles, the  $T_g$ 's proximity to body temperature makes it more likely that the particle will experience an undesired burst release of its drug payload. A  $T_g$  of PLGA nanoparticles that is around 37 °C is important to consider for drug delivery because it is close to the physiological temperature of the human body. This means that when PLGA-based drug delivery systems are administered in the body, the polymer will transition from a glassy, solid state to a rubbery, flexible state, which allows for rapid release of the drug. Notably, the  $T_g$  of DMAB-PLGA bulk is around 37 °C (Fig. 8D) even measured in dry conditions, which is significantly lower than that of PVA's (Fig. 8B). One possible explanation is that the PLGA became more liquid-like during the second heating, allowing the surface DMAB molecules to penetrate the particle matrix. Unlike PVA, which can blend with PLGA in the nanoparticles to increase the  $T_g$ , DMAB molecule with positively charged quaternary ammonium groups can be adsorbed by PLGA with the negatively charged carboxylate groups,<sup>42</sup> increasing the free volume between polymer chains and increasing polymer mobility. This plasticizing effect can make the PLGA polymer more flexible. Thus, DMAB is a small molecule that can act as a plasticizer to lower the  $T_g$ . Representative curves of the mDSC scans are plotted in Fig. 9A, and a clear  $T_g$  shift is observed from the DMAB bulk (green dotted line). Fig. 9B shows the  $\Delta T_g$  of DMAB-PLGA is significantly higher than that of PVA-PLGA.

Analysis of the  $T_g$  of PLGA nanoparticles in this study (Fig. 10A and B) also show that the  $\Delta T_g$  is smaller in water compared to dry for both sets of particles, indicating that the shift in nanoparticle  $T_g$  in water is attributable to interfacial effects in addition to plasticization, suggesting that further investigation of surfactant free PLGA nanoparticles is needed. Notably,  $\Delta T_g$  of dry DMAB samples is significantly lower compared to other  $\Delta T_g$ 's, in part because the  $T_g$  bulk of dry DMAB samples was particularly low. The possible reason is described above which is due to the plasticizing effect of DMAB.

## 4. Conclusion

The influence of surfactant on the  $T_g$  of PLGA nanoparticles was investigated using mDSC. Residual surfactant exhibited a positive correlation with initial surfactant concentration, which was also confirmed by H NMR. In  $T_g$  measurements in dry conditions, DMAB-PLGA NPs showed a significantly large  $\Delta T_g$ , notably from the decrease in  $T_g$  of the bulk, likely due to the penetration of DMAB molecules into the polymer matrix, plasticizing the sample. Additionally, a  $T_g$  reduction in wet measurement was observed, confirming that water acts as a plasticizer in PLGA. The decrease in  $T_g$  in wet conditions for

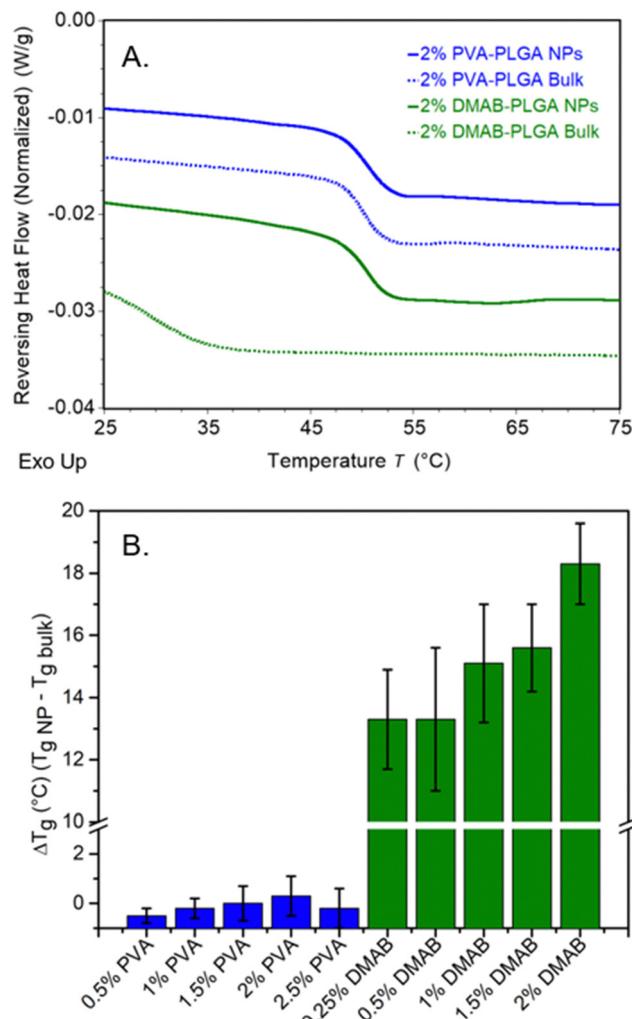


Fig. 9 Examples of mDSC plots (A) of PLGA nanoparticles (blue line) and bulk (blue dotted line) for 2% PVA-PLGA NPs (blue) and 2% DMAB-PLGA NPs (green), and (B)  $\Delta T_g$  of dry PLGA nanoparticles compared to the bulk  $T_g$  for nanoparticles made with varying amounts of PVA (blue) and DMAB (green).  $\Delta T_g$  of DMAB-PLGA nanoparticles is significantly higher than the  $\Delta T_g$  of PVA-PLGA nanoparticles (mean  $\pm$  SD,  $n = 3$ ). Curves shifted vertically for easier viewing.

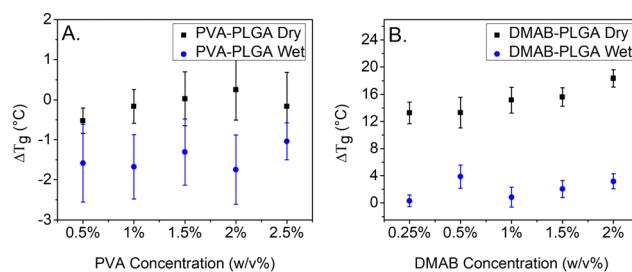


Fig. 10  $\Delta T_g$  of PLGA nanoparticles made with varying amounts of PVA (A) and DMAB (B) measured by mDSC either dry or in water.  $\Delta T_g$  represents the difference between nanoparticle  $T_g$  and bulk  $T_g$  (mean  $\pm$  SD,  $n = 3$ ).

both surfactants is noteworthy as the  $T_g$  is approaching physiological temperature, resulting in a greater burst release of drug for drug loaded particles.

Interestingly, the  $\Delta T_g$  in water was less than the  $\Delta T_g$  in dry conditions, suggesting that there are also interfacial effects from the different external environments. However, the presence of surfactant in PLGA nanoparticles complicates investigation of the interfacial effect. The difference in behavior exhibited by PLGA's  $\Delta T_g$  in comparison to other commonly studied polymers, such as PS, raise interesting questions regarding the polymer physics of PLGA in confined environments. Beyond the investigation of  $T_g$  for drug delivery applications, further study of surfactant free PLGA particles are warranted to comprehensively explain this phenomenon.

## Author contributions

K. M. conceived and directed the study. G. L. and R. M. prepared PLGA nanoparticles and measured  $T_g$ . G. L. acquire the NMR spectrums and SEM images and analyzed the resulting data. R. M. performed the intrinsic viscosity measurements. A. B. assisted with data analysis. All authors contributed to giving valuable suggestions and discussion. K. M. oversaw the project and led the research activity planning and execution.

## Conflicts of interest

There are no conflicts to declare.

## References

- 1 C. Zhang, L. Yang, F. Wan, H. Bera, D. Cun, J. Rantanen and M. Yang, *Int. J. Pharm.*, 2020, **585**, 119441.
- 2 K. Xu, N. An, H. Zhang, Q. Zhang, K. Zhang, X. Hu, Y. Wu, F. Wu, J. Xiao, H. Zhang, R. Peng, H. Li and C. Jia, *J. Drug Delivery Sci. Technol.*, 2020, **55**, 101405.
- 3 S. Shakeri, M. Ashrafizadeh, A. Zarrabi, R. Roghanian, E. G. Afshar, A. Pardakhty, R. Mohammadinejad, A. Kumar and V. K. Thakur, *Biomedicines*, 2020, **8**, 13.
- 4 C.-Y. Cheng, Q.-H. Pho, X.-Y. Wu, T.-Y. Chin, C.-M. Chen, P.-H. Fang, Y.-C. Lin and M.-F. Hsieh, *Polymers*, 2018, **10**, 519.
- 5 H. K. Makadia and S. J. Siegel, *Polymers*, 2011, **3**, 1377–1397.
- 6 L. I. Cabezas, I. Gracia, A. De Lucas and J. F. Rodríguez, *Ind. Eng. Chem. Res.*, 2014, **53**, 15374–15382.
- 7 L. Y. Lee, S. H. Ranganath, Y. Fu, J. L. Zheng, H. S. Lee, C.-H. Wang and K. A. Smith, *Chem. Eng. Sci.*, 2009, **64**, 4341–4349.
- 8 M. Alonso-Sande, A. Des Rieux, V. Fievez, B. Sarmento, A. Delgado, C. Evora, C. Remuñán-López, V. Préat and M. J. Alonso, *Biomacromolecules*, 2013, **14**, 4046–4052.
- 9 S. Fredenberg, M. Wahlgren, M. Reslow and A. Axelsson, *Int. J. Pharm.*, 2011, **415**, 34–52.
- 10 R. Karnik, F. Gu, P. Basto, C. Cannizzaro, L. Dean, W. Kyei-Manu, R. Langer and O. C. Farokhzad, *Nano Lett.*, 2008, **8**, 2906–2912.
- 11 G. Liu and K. Mcennis, *Polymers*, 2022, **14**, 993.
- 12 H. Levine and L. Slade, *Water Sci. Rev.*, 1988, **3**, 79–185.
- 13 Sonam, H. Chaudhary, V. Arora, K. Kholi and V. Kumar, *Polym. Rev.*, 2013, **53**, 546–567.
- 14 P. In Pyo Park and S. Jonnalagadda, *J. Appl. Polym. Sci.*, 2006, **100**, 1983–1987.
- 15 S. K. Sahoo, J. Panyam, S. Prabha and V. Labhasetwar, *J. Controlled Release*, 2002, **82**, 105–114.
- 16 S. Spek, M. Haeuser, M. M. Schaefer and K. Langer, *Appl. Surf. Sci.*, 2015, **347**, 378–385.
- 17 H.-Y. Kwon, J.-Y. Lee, S.-W. Choi, Y. Jang and J.-H. Kim, *Colloids Surf. A*, 2001, **182**, 123–130.
- 18 V. Bhardwaj, D. D. Ankola, S. C. Gupta, M. Schneider, C.-M. Lehr and M. N. V. R. Kumar, *Pharm. Res.*, 2009, **26**, 2495–2503.
- 19 Y. Jin, A. Xu, M. Yao, B. Li, Y. Jin and J. Ying, *Int. J. Nanomed.*, 2012, 3547, DOI: [10.2147/ijn.s32188](https://doi.org/10.2147/ijn.s32188).
- 20 R. Gossmann, K. Langer and D. Mulac, *PLoS One*, 2015, **10**, e0127532.
- 21 C. Bouissou, J. J. Rouse, R. Price and C. F. Van Der Walle, *Pharm. Res.*, 2006, **23**, 1295–1305.
- 22 J. U. Menon, S. Kona, A. S. Wadajkar, F. Desai, A. Vadla and K. T. Nguyen, *J. Biomed. Mater. Res., Part A*, 2012, **100A**, 1998–2005.
- 23 P. P. Simon and H. J. Ploehn, *J. Rheol.*, 2000, **44**, 169–183.
- 24 E. A. Dimarzio and J. H. Gibbs, *J. Polym. Sci., Part A: Gen. Pap.*, 1963, **1**, 1417–1428.
- 25 P. G. Royall, D. Q. M. Craig and C. Doherty, *Pharm. Res.*, 1998, **15**, 1117–1121.
- 26 C. Leyva-Porras, P. Cruz-Alcantar, V. Espinosa-Solís, E. Martínez-Guerra, C. I. Piñón-Balderrama, I. Compean Martínez and M. Z. Saavedra-Leos, *Polymers*, 2019, **12**, 5.
- 27 H. Rachmawati, *Sci. Pharm.*, 2016, **84**, 191–202.
- 28 J. Vysloužil, P. Doležel, M. Kejdušová, E. Mašková, J. Mašek, R. Lukáč, V. Koštál, D. Vetchý and K. Dvořáčková, *Acta Pharm.*, 2014, **64**, 403–417.
- 29 K. Miladi, S. Sfar, H. Fessi and A. Elaissari, *Ind. Crops Prod.*, 2015, **72**, 24–33.
- 30 R. Jalil and J. Nixon, *J. Microencapsulation*, 1990, **7**, 25–39.
- 31 M. Khemani, M. Sharon and M. Sharon, *Int. Scholarly Res. Not.*, 2012, **2012**, 187354.
- 32 D. L. Cooper and S. Harirforoosh, *PLoS One*, 2014, **9**, e87326.
- 33 K. Y. Hernández-Girottonini, R. J. Rodríguez-Córdova, C. A. Gutiérrez-Valenzuela, O. Peñuñuri-Miranda, P. Zavala-Rivera, P. Guerrero-Germán and A. Lucero-Acuña, *RSC Adv.*, 2020, **10**, 4218–4231.
- 34 J. Sun, J. Walker, M. Beck-Broichsitter and S. P. Schwendeman, *Drug Delivery Transl. Res.*, 2022, **12**, 720–729.
- 35 S. Khaledi, S. Jafari, S. Hamidi, O. Molavi and S. Davaran, *J. Biomater. Sci., Polym. Ed.*, 2020, **31**, 1107–1126.
- 36 P. Bonardelli, G. Moggi and A. Turturro, *Polymer*, 1986, **27**, 905–909.
- 37 L. Robeson, *Polymers*, 2014, **6**, 1251–1265.
- 38 C. Zhang, Y. Guo and R. D. Priestley, *Macromolecules*, 2011, **44**, 4001–4006.
- 39 D. Christie, C. Zhang, J. Fu, B. Koel and R. D. Priestley, *J. Polym. Sci., Part B: Polym. Phys.*, 2016, **54**, 1776–1783.
- 40 S. Feng, Z. Li, R. Liu, B. Mai, Q. Wu, G. Liang, H. Gao and F. Zhu, *Soft Matter*, 2013, **9**, 4614.
- 41 N. Passerini and D. Q. M. Craig, *J. Controlled Release*, 2001, **73**, 111–115.
- 42 T. Miyazawa, M. Itaya, G. C. Burdeos, K. Nakagawa and T. Miyazawa, *Int. J. Nanomed.*, 2021, **16**, 3937–3999.