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## Atomically accurate site-specific ligand tailoring of highly acid- and alkali-resistant Ti(IV)-based metallamacrocycle for enhanced CO<sub>2</sub> photoreduction<sup>†</sup>

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Skillfully engineering surface ligands at specific sites within robust clusters presents both a formidable challenge and a captivating opportunity. Herein we unveil an unprecedented titanium-oxo cluster: a calix [8]arene-stabilized metallamacrocycle ( $Ti_{16}L_4$ ), uniquely crafted through the fusion of four "core-shell"  $\{Ti_4@TBC[8]\}(L)$  subunits with four oxalate moieties. Notably, this cluster showcases an exceptional level of chemical stability, retaining its crystalline integrity even when immersed in highly concentrated acid (1 M HNO<sub>3</sub>) and alkali (20 M NaOH). The macrocycle's surface unveils four specific, customizable  $\mu_2$ -bridging sites, primed to accommodate diverse carboxylate ligands. This adaptability is highlighted through deliberate modifications achieved by alternating crystal soaking in alkali and carboxylic acid solutions. Furthermore,  $Ti_{16}L_4$  macrocycles autonomously self-assemble into one-dimensional nanotubes, which subsequently organize into three distinct solid phases, contingent upon the specific nature of the four  $\mu_2$ -bridging ligands. Notably, the  $Ti_{16}L_4$  exhibit a remarkable capacity for photocatalytic activity in selectively reducing CO<sub>2</sub> to CO. Exploiting the macrocycle's modifiable shell yields a significant boost in performance, achieving an exceptional maximum CO release rate of  $4.047 \pm 0.243$  mmol g<sup>-1</sup> h<sup>-1</sup>. This study serves as a striking testament to the latent potential of precision-guided surface ligand manipulation within robust clusters, while also underpinning a platform for producing microporous materials endowed with a myriad of surface functionalities.

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## Introduction

Amidst the evolving landscape of advanced materials, titania nanomaterials have garnered significant interest, particularly for their promising applications as semiconductor photocatalysts in solar energy conversion.<sup>1–3</sup> The allure of structurally precise titanium-oxo clusters (TOCs) stems from their tunable geometry and intriguing photoelectrical properties. Emerging as a captivating model, these attributes not only unravel intricate structure–property correlations within bulk TiO<sub>2</sub> phases at the atomic scale, but also offer valuable structural and

spectroscopic insights into the surface properties of TiO<sub>2</sub>.<sup>4–10</sup> Unlike conventional TiO<sub>2</sub>, TOCs exhibit a rich diversity of structural types, carrying with them a distinctive palette of traits such as stability, semiconductor-like attributes, light absorption capacity, and band structure. These features are expertly modulated by introducing various functionalized organic ligands to achieve tailored modifications.<sup>11–21</sup> Yet, the pursuit of precision in ligand modifications on specific TOCs comes with its challenges, requiring the careful negotiation of unstable coordination sites while upholding the core's inorganic integrity. Regrettably, altering protective ligands often triggers partial or complete dissection, resection, or structural reconstruction of the clusters.<sup>22,23</sup> To surmount this hurdle, the creation of exceptionally stable TOCs emerges as a pivotal prerequisite. Enter the realm of macrometallocycles (MMCs), renowned for their formidable stability rooted in robust ring structures. This inherent resilience safeguards the chemical essence of TOCs, even under rigorous conditions, paving the way for subsequent adaptations. Additionally, another approach involves enveloping the cluster core with bulky polydentate macrocyclic ligands, exemplified by calixarenes.<sup>24–26</sup> These intricate macrocycles, composed of multiple phenol

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units, have a storied history in orchestrating diverse metallo-supramolecular architectures.<sup>27–33</sup> By fusing TOCs with a macrocyclic scaffold shielded by calixarene ligands, an uncharted territory opens to explore the impact of ligand tailoring at precise sites on their physicochemical attributes. While cyclic clusters abound,<sup>34–40</sup> the realm of  $\text{Ti}^{IV}$ -based rings remains relatively untapped due to the hydrolysis susceptibility of  $\text{Ti}^{IV}$  ions. Notably, calixarene-stabilized  $\text{Ti}^{IV}$ -MMCs have yet to grace the literature. The envisioned assembly of atomically precise MMCs of  $\text{Ti}^{IV}$ , secured by calixarene guardians, holds immense promise for ushering rational design and optimizing performance across future applications.

Herein, an exceptionally stable *p*-*tert*-butylcalix[8]arene-protected  $\text{Ti}^{IV}$ -based metallamacrocycles (MMC), denoted as  $[\text{H}_4\text{Ti}_{16}\text{O}_8(\text{TBC}[8])_4(\text{Oa})_4(\text{Ac})_4(\text{iPrO})_8]$ , was synthesized. This MMC showcases remarkable resilience, displaying resistance against an array of organic solvents, concentrated acid (1 M  $\text{HNO}_3$ ), and alkali (20 M  $\text{NaOH}$ ), thus establishing itself as a paramount example of cluster stability. The gigantic cluster has a “donut” shape with an inner diameter of 12 Å, an outer diameter of 30 Å, and a height of 18 Å, making it the largest known TOC in the metal-calixarene system. A pivotal discovery lies in the identification of four modifiable coordination sites on the  $\text{Ti}_{16}\text{L}_4$  surface, which has led to further exploration of its exchange activity and applications (Scheme 1). These sites, intriguingly, offer two distinct avenues for functionalization: (1) utilizing a one-pot synthesis method, the  $\mu_2$ -bridging sites can be occupied by acetate and  $\text{Cl}^-$ . The ratio of these ligands can be precisely tailored from 4:0 to 2:2, and further to 0:4. Noteworthy is the emergence of a 3D network in  $\text{Ti}_{16}\text{L}_4$ -3, modified with four  $\text{Cl}^-$ , yielding infinite cylindrical channels with diameters of approximately 2 nm. (2) An alternative post-modification strategy involves a sequential immersion of the cluster in concentrated alkali and carboxylic acid solutions, facilitating reversible exchange of the  $\mu_2$ -bridging sites by  $\text{OH}^-$  and diverse carboxylates. The molecular-level understanding of this exchange process is unveiled through crystal-to-crystal diffraction studies, shedding light on structural transformations and ligand dynamics. Significantly, modifying the  $\text{Ti}_{16}\text{L}_4$  shell distinctly influences physicochemical attributes, encompassing photocurrent response, hydrophilicity and energy levels, with

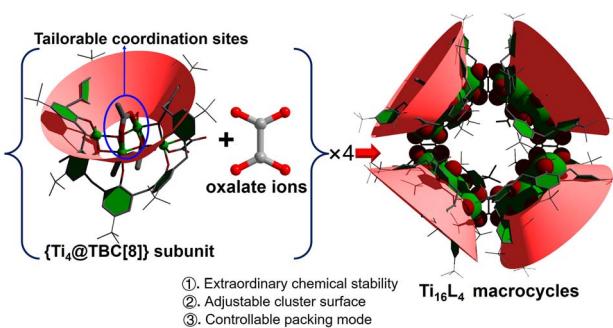
ultimate implications for the photocatalytic  $\text{CO}_2$  reduction potential of the clusters. The ability to organically modify the shell while preserving macrocycle integrity introduces a pioneering avenue for fine-tuning chemical properties, setting the stage for these promising  $\text{Ti}^{IV}$ -based MMCs to be harnessed across diverse applications.

## Results and discussion

The synthesis of  $\text{Ti}_{16}\text{L}_4$ -1 entailed a one-pot solvothermal reaction involving  $\text{TBC}[8]$ , oxalic acid ( $\text{H}_2\text{Oa}$ ), and  $\text{Eu}(\text{Ac})_3$  in conjunction with  $\text{Ti}(\text{O}^{\text{i}}\text{Pr})_4$ , conducted within acetonitrile ( $\text{CH}_3\text{CN}$ ) at 120 °C for 72 hours. This procedure yielded red crystals with a notable 65% yield and consistent reproducibility. The ensuing single-crystal X-ray diffraction (SCXRD) analysis unveiled  $\text{Ti}_{16}\text{L}_4$ -1's monoclinic system crystallization in space group  $C2/c$ .<sup>41</sup> The composition was precisely identified as  $[\text{H}_4\text{Ti}_{16}\text{O}_8(\text{TBC}[8])_4(\text{Oa})_4(\text{Ac})_4(\text{iPrO})_8]$ . X-ray photoelectron spectroscopy (XPS) analysis further validated the presence of  $\text{Ti}^{IV}$ . The  $\text{Ti}$  2p spectrum exhibited distinct regions at  $\text{Ti} 2\text{p}_{1/2}$  (*ca.* 464.9 eV) and  $\text{Ti} 2\text{p}_{3/2}$  (*ca.* 459.0 eV) with a binding energy separation of 5.9 eV, confirming the existence of  $\text{Ti}^{IV}$  (Fig. S32†). A comprehensive structural analysis elucidated the intricate architecture of the cluster. It emerged as a “core–shell” assembly comprising four  $\{\text{Ti}_4@\text{TBC}[8]\}$  subunits interconnected by four oxalate ligands (Fig. 1A and B). Within each  $\{\text{Ti}_4@\text{TBC}[8]\}$  subunit, four  $\text{Ti}^{IV}$  ions formed a central  $\{\text{Ti}_4\text{O}_2\}$  core connected by  $\mu_3\text{O}^{2-}$  ions, encased within the  $\text{TBC}[8]$  cavity (Fig. 1C). The  $\text{TBC}[8]$  molecule exhibits two distinct bonding modes for its eight phenoxide groups. Six of these groups are individually bonded to one  $\text{Ti}^{IV}$ , while the remaining two groups are bridged by two  $\text{Ti}^{IV}$  in a  $\mu_2\text{-K}^2(\text{O})$  fashion. The  $\{\text{Ti}_4\text{O}_2\}$  core showcased two distinct sets of  $\text{Ti}^{IV}$  sites. The  $\text{Ti}(2)$  and  $\text{Ti}(3)$  sites possess identical coordination environments, each coordinated by two phenol O, two  $\mu_3\text{O}^{2-}$  ions, and one  $\text{iPrO}^-$  group. An additional acetate linked these two  $\text{Ti}^{IV}$  in a  $\mu_2\text{-}(\text{O},\text{O}')$  bridging fashion. Additionally,  $\text{Ti}(1)$  and  $\text{Ti}(4)$  exhibited equivalent coordination environments immobilized within the  $\text{TBC}[8]$  cavity.  $\text{TBC}[8]$ 's flexibility allowed the attachment of an  $\text{Oa}^{2-}$  ligand to  $\text{Ti}(1)$  and  $\text{Ti}(4)$ , while the oxalate ligand adopted a  $\mu_2\text{-}(\text{O},\text{O}',\text{O}')$  mode, bridging two  $\{\text{Ti}_4@\text{TBC}[8]\}$  subunits into a dimeric structure. This unique oxalate binding played a pivotal role in forming the closed-loop structure. Four oxalate ligands vertically bridged four  $\{\text{Ti}_4@\text{TBC}[8]\}$  units, culminating in a quadrilateral macrocycle characterized by a 90° internal angle (Fig. 1D). Notably, this macrocycle exhibited an outer diameter of 30 Å and a height of 18 Å, with internal voids measuring an inner diameter of 12 Å and a window size of 7.5 Å, all determined by structural features.

### Ligand modification

**One-spot method.** As demonstrated in prior studies, the unique structural motifs of the four  $\{\text{Ti}_4@\text{TBC}[8]\}$  subunits encompass embellished  $\text{Ti}_2$  and  $\text{Ti}_3$  sites, each adorned with a single acetate and two weakly coordinated  $\text{iPrO}^-$  groups. Intriguingly, through a deliberate alteration involving the



Scheme 1 Illustration of the assembly of the  $\text{Ti}_{16}\text{L}_4$  macrocycle with specific tailorabile sites.



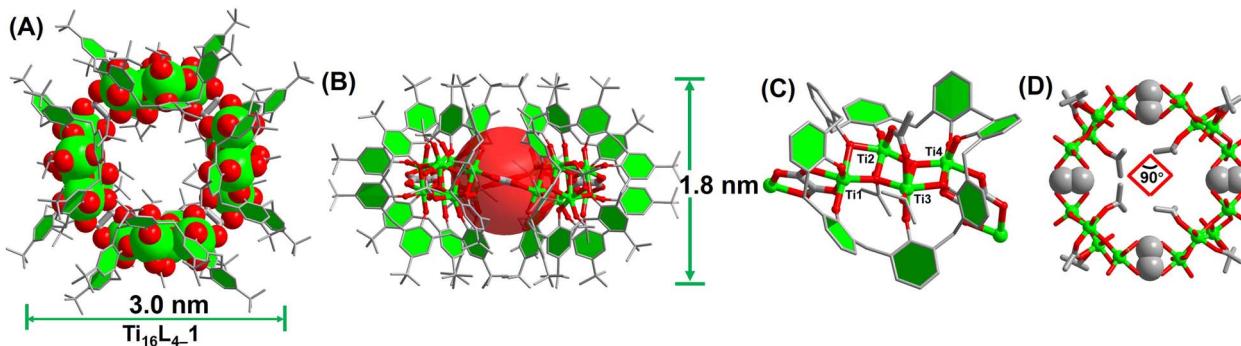


Fig. 1 Structural depictions of  $\text{Ti}_{16}\text{L}_{4-1}$ . (A) and (B) Overhead and lateral perspectives; (C) configuration of the  $\{\text{Ti}_4@\text{TBC}[8]\}$  subunit; (D) arrangement of the  $\{\text{Ti}_{16}\}$  nanoring.

substitution of  $\text{Eu}(\text{Ac})_3$  with a judicious blend of  $\text{Eu}(\text{Ac})_3$  and  $\text{EuCl}_3$ , or by sequential addition of  $\text{EuCl}_3$  during the synthetic process, distinct crystalline entities,  $\text{Ti}_{16}\text{L}_{4-2}$  and  $\text{Ti}_{16}\text{L}_{4-3}$ , are successfully synthesized.<sup>42,43</sup> For  $\text{Ti}_{16}\text{L}_{4-2}$ , the architectural formulation reads  $[\text{H}_4\text{Ti}_{16}\text{O}_8(\text{TBC}[8])_4(\text{Oa})_4(\text{Ac})_2\text{Cl}_2(\text{iPrO})_8]$ , while  $\text{Ti}_{16}\text{L}_{4-3}$  is denoted by the composition  $[\text{H}_4\text{Ti}_{16}\text{O}_8(\text{TBC}[8])_4(\text{Oa})_4\text{Cl}_4(\text{iPrO})_8]$ . Remarkably, these two clusters mirror the macrocyclic framework of  $\text{Ti}_{16}\text{L}_{4-1}$ , while diverging in their surface ligands. A notable feature is the dynamic occupancy of the four  $\mu_2$ -bridging sites within the macrocycles, where acetate and  $\text{Cl}^-$  find a variable equilibrium. The precise modulation of this ligand interplay is evident, transitioning from a 4 : 0 ratio in  $\text{Ti}_{16}\text{L}_{4-1}$ , to a balanced 2 : 2 ratio in  $\text{Ti}_{16}\text{L}_{4-2}$  (Fig. 2A), culminating in an exclusive 0 : 4 ratio in  $\text{Ti}_{16}\text{L}_{4-3}$  (Fig. 2B).

Beyond the intricate geometries that clusters adopt, the interplay between surface ligands emerges as a pivotal determinant of their crystalline packing and consequential material attributes. Mastery over these multifaceted factors remains a formidable task. Our study, however, illuminates a path to precision control in this arena. Strikingly, while the shared macrocyclic scaffold remains constant, variations in surface ligands orchestrate disparate packing configurations within the crystal lattice of  $\text{Ti}_{16}\text{L}_{4-1/2/3}$ . Within all phases, the metallacycles ingeniously arrange face-to-face, thus sculpting 1D nanotubes that harmonize through van der Waals forces, culminating in their assemblage (Fig. 2C and D). The meticulous orchestration of this assembly begins by staggering neighboring nanotubes to alleviate steric hindrance stemming

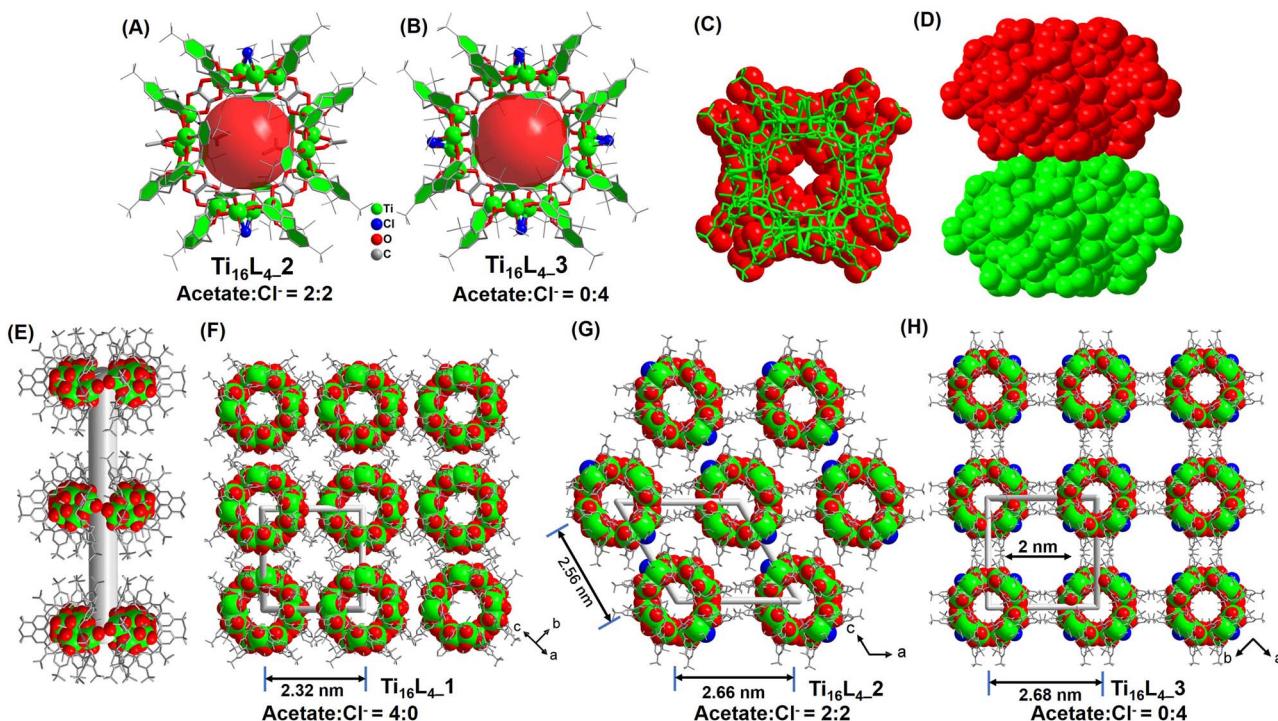


Fig. 2 (A) and (B) Configuration of the  $\text{Ti}_{16}\text{L}_{4-2/3}$  macrometallocycles (MMCs); (C) and (D) proximity arrangement of two  $\text{Ti}_{16}\text{L}_4$  macrocycles; (E) formation of the nanotube within  $\text{Ti}_{16}\text{L}_4$ ; (F)–(H) three distinct patterns of nanotube arrangement, including square and hexagonal arrays.



from *tert*-butyl moieties of adjacent macrocycles (Fig. 2E). This choreographed dance then evolves into a parallel stacking, fashioning an assorted array of 3D architectures. Remarkably, **Ti<sub>16</sub>L<sub>4</sub>-1**'s nanotubes organize into a square matrix (Fig. 2F), whereas **Ti<sub>16</sub>L<sub>4</sub>-2**'s nanotubes adopt a hexagonal pattern (Fig. 2G). This intriguing divergence traces its origins back to the precise occupancy of the four  $\mu_2$ -bridging sites. The substitution of acetate with Cl<sup>-</sup> profoundly impacts C-H $\cdots$ C-H interactions among the clusters. Additionally, the presence of polar Ti-Cl bonds endows the surface of **Ti<sub>16</sub>L<sub>4</sub>-2**, now embellished with two Cl<sup>-</sup>, with a discernible negative charge. This newfound electrostatic repulsion between the nanotubes impels their separation within the lattice. Evidently, the centroid distance between adjacent nanotubes escalates from 2.32 nm (as observed in **Ti<sub>16</sub>L<sub>4</sub>-1**) to 2.56 and 2.66 nm in succession. This captivating phenomenon finds further exemplification in the assembly motif of **Ti<sub>16</sub>L<sub>4</sub>-3**, wherein all four  $\mu_2$ -bridging sites embrace Cl<sup>-</sup>, ensuing in yet another square arrangement of macrocycles. Notably, a subtle 45° rotation between neighboring nanotubes in **Ti<sub>16</sub>L<sub>4</sub>-1** mitigates the dominant electrostatic repulsion, paving the way for the emergence of 1D channels along the *c*-axis. These channels, a noteworthy 2  $\times$  2 nm in dimension, define **Ti<sub>16</sub>L<sub>4</sub>-3**'s unique character (Fig. 2H). The resplendent novelty of these findings lies in the precise manipulation of ligand dynamics to orchestrate such diverse and intricate packing phenomena.

The intricate interlocking of Ti<sub>16</sub>L<sub>4</sub> macrocycles intricately assembles into meticulously ordered nanotubular frameworks across all three distinct phases. To validate the porosity of these architectures, we conducted N<sub>2</sub> adsorption/desorption analyses that unveiled characteristic type I isotherms for each phase, unmistakably indicating their inherent porous nature. Concretely, our evaluation yielded Brunauer–Emmett–Teller (BET) surface areas of 209.24 m<sup>2</sup> g<sup>-1</sup> for **Ti<sub>16</sub>L<sub>4</sub>-1**, 389.92 m<sup>2</sup> g<sup>-1</sup> for **Ti<sub>16</sub>L<sub>4</sub>-2**, and a remarkable 813.46 m<sup>2</sup> g<sup>-1</sup> for **Ti<sub>16</sub>L<sub>4</sub>-3** (Fig. 3A). The underlying absorbance potential was corroborated by diffuse reflectance spectra, revealing robust light absorption in the visible spectrum (Fig. S39†). Capitalizing on this light absorption attribute, we ventured into assessing the

visible-light-driven photocurrent responses of Ti<sub>16</sub>L<sub>4</sub>. Through repeated irradiation cycles, steady photocurrent responses were uniformly observed across all clusters. Notably, the substitution of acetate with Cl<sup>-</sup> yielded an enhancement in photocurrent density. Among the clusters, **Ti<sub>16</sub>L<sub>4</sub>-3** emerged as the standout, demonstrating the most optimal photocurrent response. Its recorded current density surged to an impressive 3.2  $\mu$ A cm<sup>-2</sup>, which was approximately threefold higher than that of **Ti<sub>16</sub>L<sub>4</sub>-1** (1.0  $\mu$ A cm<sup>-2</sup>) (Fig. 3B). Interestingly, the hydrophobicity of Ti<sub>16</sub>L<sub>4</sub> crystals emerged as an outcome intricately linked to their specific stacking arrangements. Evidently, water droplets exhibited distinct behaviors on the surfaces of **Ti<sub>16</sub>L<sub>4</sub>-1/2** powders, maintaining rounded shapes with contact angles measuring 138.5° and 117.5°, respectively (Fig. 3C). In a striking departure, the contact angle observed for **Ti<sub>16</sub>L<sub>4</sub>-3** was notably reduced to a mere 49.5°, indicative of its pronounced hydrophilic character. This intriguing contrast owes its origins to the presence of polar Ti-Cl bonds and the substantial presence of expansive cylindrical channels within the structural matrix, collectively orchestrating this distinctive behavior.

The stability of the nanocluster emerges as a pivotal facet, particularly in the context of post-synthetic modifications, warranting meticulous consideration. Significantly, the calixarene modification of the Ti<sub>16</sub>L<sub>4</sub> confers a profound enhancement in chemical stability. Evidently, our PXRD analysis substantiated the robust stability of the **Ti<sub>16</sub>L<sub>4</sub>-1** crystal across an array of solvents such as toluene, methanol, CH<sub>3</sub>CN, and DMF (Fig. S26†). To further probe its solution behavior, we harnessed matrix-assisted laser desorption/ionization time-of-flight mass spectrometry (MALDI-TOF-MS) on **Ti<sub>16</sub>L<sub>4</sub>-1** dissolved within a mixed solvent of CHCl<sub>3</sub> and MeOH (Fig. 4A). Encouragingly, the dominant peaks in the spectrum unequivocally corresponded to species like [H<sub>4</sub>Ti<sub>16</sub>O<sub>8</sub>(TBC[8])<sub>4</sub>(Oa)(-Ac)<sub>4</sub>(<sup>i</sup>PrO)<sub>x</sub>(MeO)<sub>y</sub>]<sup>+</sup> ( $x + y = 7$ ). This finding underscores that, while retaining the overall architecture of Ti<sub>16</sub>L<sub>4</sub>, the <sup>i</sup>PrO<sup>-</sup> on the cluster exhibit high exchangeability with MeO<sup>-</sup> ions, while the chelating acetate sites remain resolutely stable. Dynamic

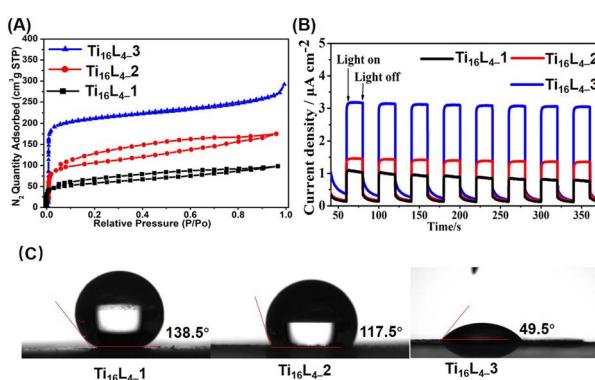


Fig. 3 (A) N<sub>2</sub> adsorption/desorption isotherm recorded at 77 K; (B) transient photocurrent responses of Ti<sub>16</sub>L<sub>4</sub> captured during Xe lamp irradiation in a 0.2 M Na<sub>2</sub>SO<sub>4</sub> electrolyte solution; (C) comparative assessment of the wettabilities across **Ti<sub>16</sub>L<sub>4</sub>-1**, **Ti<sub>16</sub>L<sub>4</sub>-2** and **Ti<sub>16</sub>L<sub>4</sub>-3**.

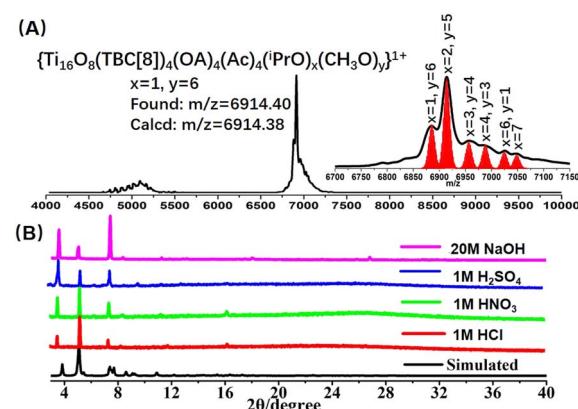


Fig. 4 (A) Positive-mode MALDI-TOF-MS spectrum depicting **Ti<sub>16</sub>L<sub>4</sub>-1** in CHCl<sub>3</sub> solution; (B) PXRD patterns of **Ti<sub>16</sub>L<sub>4</sub>-1** acquired during acid- and base-stability assessments. Remarkably, **Ti<sub>16</sub>L<sub>4</sub>-1** exhibits robustness even after prolonged exposure to 1 M HCl, HNO<sub>3</sub>, H<sub>2</sub>SO<sub>4</sub>, and 20 M NaOH for 24 hours.



light scattering measurements further confirmed a singular size distribution with an average diameter of 4.5 nm, a fitting match to the cluster size, concretely affirming the presence of well-defined  $\text{Ti}_{16}\text{L}_4$  entities in solution (Fig. S37†). Another distinctive feature of  $\text{Ti}_{16}\text{L}_4\text{-1}$  crystals comes to the fore in the realm of water stability—an attribute often fraught with challenges in many cluster crystals. In stark contrast, the exceptional pH stability of  $\text{Ti}_{16}\text{L}_4\text{-1}$  crystals prevails over a broad pH values spanning from 1 to 14 (Fig. S27†). Subsequent testing reinforces this extraordinary stability, with the crystals remaining unaltered even in the face of concentrated acids (1 M HCl, 1 M  $\text{H}_2\text{SO}_4$ , and 1 M  $\text{HNO}_3$ ) and alkali (20 M NaOH) over a 24 hour period (Fig. 4B). This robust endurance finds its basis in the hydrophobic calixarene outer chamber enveloping  $\text{Ti}_{16}\text{L}_4$ , serving as an effective barrier against acid/alkali corrosion of the hydrophilic  $\text{Ti}^{4+}$ . This ingeniously designed spatial protection imparts a remarkably high level of water stability to the cluster.

**Post-synthetic modification.** Through an in-depth analysis of the  $\text{Ti}_{16}\text{L}_4\text{-1}$  structure, we find that it does present  $4 \times 3 = 12$  labile coordination sites, comprising four  $\mu_2$ -acetate sites and eight weakly coordinated  $^{\text{t}}\text{PrO}^-$  sites. Complementary mass spectrometry experiments have intriguingly demonstrated the dynamic nature of  $^{\text{t}}\text{PrO}^-$  in solution, allowing rapid replacement by  $\text{MeO}^-$ . Eager to bolster the surface functionality of  $\text{Ti}_{16}\text{L}_4$ , our focus shifted towards investigating the substitution of the four  $\mu_2$ -acetate sites through post-synthetic modification. The exceptional stability of  $\text{Ti}_{16}\text{L}_4$  crystals offered us an opportunity to pinpoint the exchange sites at the molecular level, employing the powerful SCXRD techniques that provided nuanced insights into the intricacies of the ligand exchange process (Fig. 5A). Embarking on a journey of formic acid (HFa) immersion, a carefully orchestrated sequence unveiled fascinating revelations. SCXRD scrutiny after immersing the  $\text{Ti}_{16}\text{L}_4\text{-1}$  crystal in 1 M formic acid for 12 h highlighted the replacement of only the four innermost labile  $^{\text{t}}\text{PrO}^-$  sites within the ring,

now replaced by  $\text{H}_2\text{O}$  molecules (yielding  $\text{Ti}_{16}\text{L}_4\text{-1(a)}$ ,<sup>44</sup> Fig. 5B). Elevated HFa concentration (5 M) and prolonged soaking (24 h) triggered further substitution, this time of the four acetate sites by formate ligands (resulting in  $\text{Ti}_{16}\text{L}_4\text{-1/HFa(a)}$ , Fig. 5C). Notably, the stability of the  $\mu_2\text{-}(\text{O},\text{O}')$  chelating form conferred a thermodynamically challenging nature to the exchange of these four acetate sites, mandating a high-concentration, extended duration immersion for successful replacement. In parallel, the terminal  $^{\text{t}}\text{PrO}^-$  sites, inherently labile, seamlessly surrendered to  $\text{H}_2\text{O}$  molecules, while the external  $^{\text{t}}\text{PrO}^-$  sites, shrouded within the hydrophobic outer calixarene cavity, remained impervious to  $\text{H}_2\text{O}$  access. An interesting color transformation emerged from the immersion of  $\text{Ti}_{16}\text{L}_4\text{-1}$  crystals in 5 M NaOH, with a rapid shift from red to yellow within 30 minutes. SCXRD analysis of the resulting yellow crystal (now  $\text{Ti}_{16}\text{L}_4\text{-1/NaOH}$ ,<sup>45</sup> Fig. 5D) illuminated a thorough overhaul—complete replacement of all four acetates and eight  $^{\text{t}}\text{PrO}^-$  ligands by  $\text{OH}^-$ , bolstered by numerous  $\text{Na}^+$  that connected through  $\text{OH}^-$  bridging. Subsequent immersion in 1 M HFa for 30 minutes not only restored the red hue but also the crystalline phase (now  $\text{Ti}_{16}\text{L}_4\text{-1/HFa(b)}$ ).<sup>46</sup> SCXRD verification elucidated the reintroduction of the four  $\mu_2$ -sites *via* formate ions, reverting to the  $\mu_2\text{-}(\text{O},\text{O}')$  configuration (Fig. 5E). Encouragingly, this intricate exchange process proved fully reversible, evidenced by the cycle of color transformation—the red crystals of  $\text{Ti}_{16}\text{L}_4\text{-1/HFa(b)}$  reverting to yellow during re-immersion in 5 M NaOH, indicative of formate ions being replaced by  $\text{OH}^-$  ions. Impressively, even after undergoing five alternating cycles of immersion in 1 M HFa and 5 M NaOH, the crystal retained its structural stability (Fig. S28†). Remarkably, these color shifts in response to acid and alkali serve as tangible indicators of the comprehensiveness of the ligand exchange process—a visually compelling testament to the groundbreaking strides of our work.

Drawing from the above insightful findings, the substitution of the four  $\mu_2$ -acetate sites within  $\text{Ti}_{16}\text{L}_4$  emerges as a dynamic

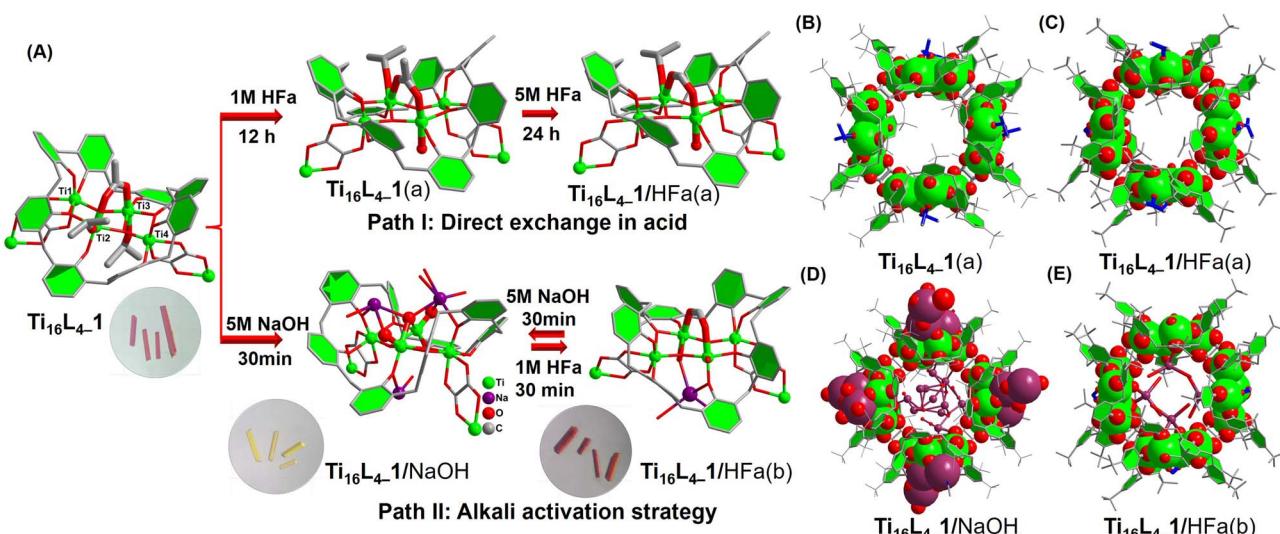


Fig. 5 (A) Schematic representation of two different post-modification paths: (B–E) crystal structures of the derived clusters of  $\text{Ti}_{16}\text{L}_4\text{-1}$ .

process, amenable to two distinct pathways. The first, path I, involves the direct immersion of crystals in a concentrated formic acid solution for a minimum of 24 hours. The second, path II, unfolds as a sequential immersion, commencing with a solution of NaOH, followed by a carboxylic acid. This latter path capitalizes on acid–base neutralization reactions, rendering the exchange process swift, completed within a mere hour. Evaluating the efficacy of these approaches, we embarked on a mission to extend the carboxylate modifications on the  $\text{Ti}_{16}\text{L}_4$  surface. Encouragingly, the gamut encompassed successful modifications: chloroacetate ( $\text{Cl}^-$ ) and bromoacetate ( $\text{Br}^-$ ) through path I, and acetate ( $\text{Ac}^-$ ),<sup>47</sup> aminoacetate ( $\text{Ama}^-$ ), glycinate ( $\text{Ga}^-$ ),<sup>48</sup> and oxalate ( $\text{Oa}^{2-}$ )<sup>49</sup> via path II. A comprehensive SCXRD analysis discerned the occupation of all four sites by these carboxylates. Delving into specifics, the bridging modes of  $\text{Ac}^-$ ,  $\text{Ama}^-$ ,  $\text{Cl}^-$ , and  $\text{Br}^-$  ligands aligned uniformly, establishing  $\mu_2\text{-}(\text{O},\text{O}')$  bridges connecting two Ti sites (as depicted in Fig. 6A–D). In marked contrast,  $\text{Oa}^{2-}$  and  $\text{Ga}^-$  ligands exhibited an intriguingly different binding pattern (as showcased in Fig. 6E and F), with the ligands forming a chelating arrangement around the  $\text{Ti}_3$  site in a bidentate  $\mu_1\text{-}(\text{O},\text{O})$  mode, concomitantly coordinating the  $\text{Ti}_2$  site with two  $\text{OH}^-$  ions. Notably, the structural analysis of  $\text{Ti}_{16}\text{L}_4\text{-1}$  modified with trifluoroacetate ( $\text{Tfa}^-$ ) encountered challenges due to poor crystal quality. This phenomenon can be attributed to the introduction of a significant number of F atoms to the cluster's surface, engendering repulsion among clusters within the tetragonal phase and leading to lattice displacement. However, this issue was effectively surmounted in the case of the  $\text{Tfa}^-$  modified  $\text{Ti}_{16}\text{L}_4\text{-2}$  cluster, revealing a definitive structural determination within the hexagonal lattice of  $\text{Ti}_{16}\text{L}_4\text{-2}$  (Fig. 6G), underscoring the adaptability of the  $\text{Tfa}^-$  to occupy these sites.

Interestingly, our explorations extended to the realm of inorganic oxyanions. When  $\text{Ti}_{16}\text{L}_4\text{-1}/\text{NaOH}$  crystals were immersed in 1 M  $\text{H}_3\text{PO}_4$  for 30 minutes, red crystals of  $\text{Ti}_{16}\text{L}_4\text{-1}/\text{H}_3\text{PO}_4$  were obtained, retaining crystallinity. SCXRD analysis brought forth an intriguing revelation— $\text{PO}_4^{3-}$  did not interact with the clusters. Instead,  $\text{OH}^-$  persisted as occupants within the 12 labile sites, with  $\text{Na}^+$  ions being entirely dislodged from the structure. Notably, the cumulative occupancy rate of the

four  $\mu_2\text{-}\text{OH}^-$  sites reached 2.80 (Fig. S22†). This intriguing outcome underscores our capacity to effect partial ligand removal from these four  $\mu_2$ -sites through a meticulous interplay of two acid–base neutralization reactions. This multifaceted exploration, together with the systematic elucidation of intricate ligand exchange pathways, reflects the groundbreaking dimension of our research endeavor.

## CO<sub>2</sub> photoreduction activities

The promising avenue of photocatalytically reducing CO<sub>2</sub> into reusable chemicals looms large as a critical step toward carbon neutrality.<sup>50</sup> However, the reported TOCs for driving such reactions remain limited, evoking a pressing need for innovative solutions.<sup>51–55</sup> Harnessing the unparalleled stability, controllable ligands, and adaptable packing modes inherent to  $\text{Ti}_{16}\text{L}_4$ , we ventured into their application as heterogeneous photocatalysts for CO<sub>2</sub> reduction. Conceiving the intricate setup, CO<sub>2</sub> photoreduction over  $\text{Ti}_{16}\text{L}_4$  unfolded under visible light irradiation ( $\lambda \geq 420$  nm) in the presence of  $[\text{Ru}(\text{bpy})_3]\text{Cl}_2 \cdot 6\text{H}_2\text{O}$  as a photosensitizer, triethanolamine (TEOA) as a sacrificial agent, and a  $\text{CH}_3\text{CN}/\text{H}_2\text{O}$  (4 : 1, v/v) solvent composition. A comprehensive analysis of the product, employing gas chromatography (GC, Fig. S47†), ion chromatography (IC, Fig. S51†) and <sup>1</sup>H NMR (Fig. S52†), lucidly affirmed the exclusive generation of CO via CO<sub>2</sub>RR, with a marginal presence of H<sub>2</sub>. It's noteworthy that no products were observed in the absence of irradiation, CO<sub>2</sub> or clusters, or the yields were exceptionally low (Table S4†). This unequivocally underscores the orchestrated interplay of each component within the photocatalytic system. Comparative assessment of the catalytic activity of  $\text{Ti}_{16}\text{L}_4\text{-1/2/3}$  in CO<sub>2</sub> photoreduction revealed a remarkable hierarchy in CO yields.  $\text{Ti}_{16}\text{L}_4\text{-1}$  produced a modest CO yield of  $11.97 \pm 0.67 \mu\text{mol}$  within 5 hours, with  $\text{Ti}_{16}\text{L}_4\text{-2}$  and  $\text{Ti}_{16}\text{L}_4\text{-3}$  exhibiting substantially enhanced CO yields of  $29.97 \pm 2.17 \mu\text{mol}$  and  $60.70 \pm 3.65 \mu\text{mol}$ , respectively (Fig. 7A).  $\text{Ti}_{16}\text{L}_4\text{-3}$  emerged as the frontrunner in catalytic prowess, displaying an impressive CO production rate of  $4046.67 \pm 243.33 \mu\text{mol g}^{-1} \text{h}^{-1}$ , coupled with an exceptional selectivity of 96.28%. These activity and selectivity levels surpassed those of many cluster or MOF-based

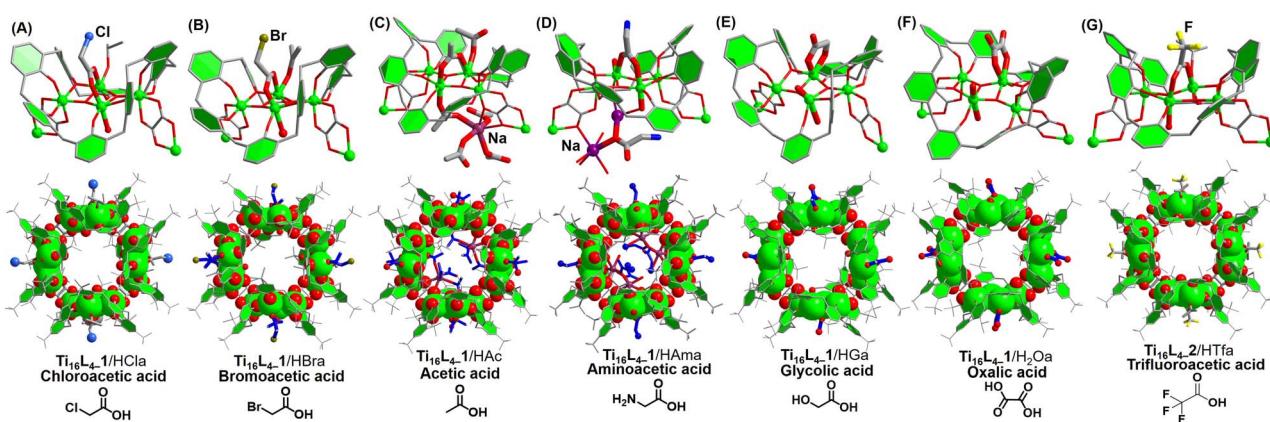


Fig. 6 (A)–(G) Crystal structures of modified clusters derived from  $\text{Ti}_{16}\text{L}_4$  via post-modification pathways.



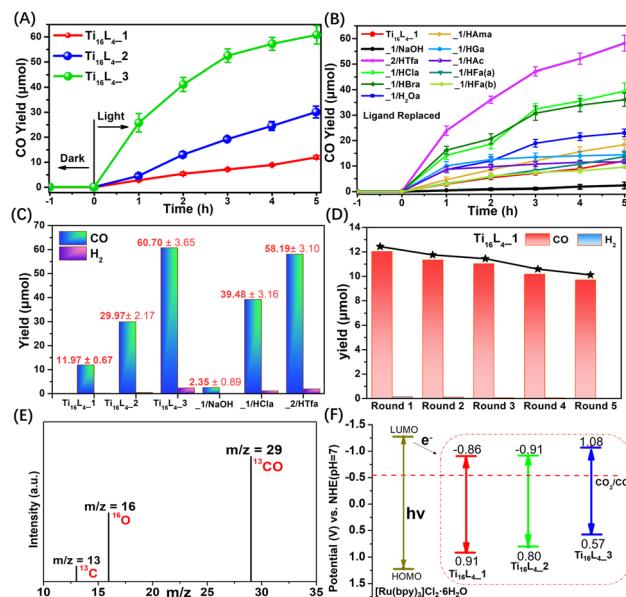


Fig. 7 (A) Time-dependent CO generation profiles for  $\text{Ti}_{16}\text{L}_4\text{-1/2/3}$ ; (B) time-dependent CO generation profiles for  $\text{Ti}_{16}\text{L}_4$  with various ligand modifications; (C) CO production rates over  $\text{Ti}_{16}\text{L}_4$ ; (D) durability assessments of  $\text{Ti}_{16}\text{L}_4\text{-1}$ . (E)  $^{13}\text{CO}$  recorded under  $^{13}\text{CO}_2$  atmosphere; (F) the energy band structure diagram for  $\text{Ti}_{16}\text{L}_4$ . Reaction conditions: 3 mg of  $\text{Ti}_{16}\text{L}_4$ , 10 mg  $[\text{Ru}(\text{bpy})_3]\text{Cl}_2\cdot 6\text{H}_2\text{O}$ , 5 mL triethanolamine (TEOA), acetonitrile (8 mL),  $\text{H}_2\text{O}$  (2 mL),  $\text{CO}_2$  (1 atm),  $\lambda \geq 420$  nm, 25 °C, 5 h.

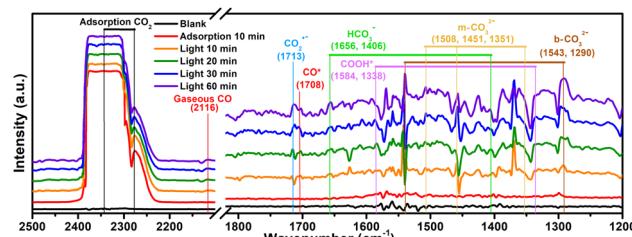
materials for  $\text{CO}_2$  to CO conversion.<sup>56–61</sup> The catalytic proficiency of this macrocyclic series aligns with the potentially catalytically active  $\text{Ti}^{\text{IV}}$  sites boasting flexible coordination spaces (specifically the Ti2 and Ti3 sites), which stand poised for  $\text{CO}_2$  adsorption. In  $\text{Ti}_{16}\text{L}_4\text{-3}$ ,  $\text{Cl}^-$  exhibit a greater tendency to vacate compared to acetates in  $\text{Ti}_{16}\text{L}_4\text{-1}$ . Consequently, the Ti sites in  $\text{Ti}_{16}\text{L}_4\text{-3}$  can showcase higher catalytic activity. This assessment of catalytic activity, supported by Table S5,† revealed that each potential  $\text{Ti}^{4+}$  catalytic site (TON<sub>Ti</sub>) in  $\text{Ti}_{16}\text{L}_4\text{-3}$  exhibited higher activity than those in  $\text{Ti}_{16}\text{L}_4\text{-1/2}$ . Additionally, the strategic advantage of the 2 nm macrochannel within  $\text{Ti}_{16}\text{L}_4\text{-3}$  unveiled a larger surface area, fostering heightened  $\text{CO}_2$  adsorption and exposure to additional catalytic enclaves. Experimental findings in  $\text{CO}_2$  adsorption bolstered these revelations (Fig. S36†), where  $\text{Ti}_{16}\text{L}_4\text{-3}$  showcased superior performance over  $\text{Ti}_{16}\text{L}_4\text{-1/2}$ , with  $\text{CO}_2$  uptake values of 57.38 and 48.79  $\text{cm}^3 \text{g}^{-1}$ , respectively, at 273 K and 298 K under 1 bar pressure. Intriguingly, functionalization of ligands on  $\text{Ti}_{16}\text{L}_4$  also had a notable influence on catalytic activity (Fig. 7B). In the case of  $\text{Ti}_{16}\text{L}_4\text{-1/NaOH}$ , all potential catalytic sites on the cluster's surface are occupied by  $\text{OH}^-$  ions. Due to the presence of strong Ti-OH bonds,  $\text{CO}_2$  is unable to access the catalytic Ti sites, resulting in a decrease in catalytic activity ( $2.35 \pm 0.89 \mu\text{mol}$  for 5 hours). Noteworthy examples include  $\text{Ti}_{16}\text{L}_4\text{-1/HBra}$  and  $\text{Ti}_{16}\text{L}_4\text{-1/HCl}$ , unveiling CO yields of  $36.25 \pm 2.59 \mu\text{mol}$  and  $39.48 \pm 3.16 \mu\text{mol}$ , respectively, after 5 hours, approximately threefold higher than pristine  $\text{Ti}_{16}\text{L}_4\text{-1}$  (Fig. 7C). Similarly,  $\text{Ti}_{16}\text{L}_4\text{-2/HTfa}$  exhibited double the CO production of its untouched counterpart ( $58.18 \pm 3.10 \mu\text{mol}$  for 5 hours),  $\text{Ti}_{16}\text{L}_4\text{-2}$ . The higher photocatalytic performance of those derived clusters is mainly due to their

modification with halogenated carboxylates that have the ability to quickly transfer charges. These enhancements find resonance in the modification-induced shifts within the cluster's bandgap structures, catalyzing accelerated electron transfers, and amplifying catalytic efficiency.

Collectively, these revelations converge to spotlight the photocatalytic acumen inherent within the  $\text{Ti}_{16}\text{L}_4$  macrocycles, unfurling their supremacy in  $\text{CO}_2$  reduction and accentuating their proclivity for selectively generating CO. What's more, the strategic orchestration of the  $\text{Ti}_{16}\text{L}_4$ 's spatial configuration, coupled with tailored enhancements infused into its outer ligands, emerges as a potent avenue for catapulting its catalytic performance to new heights. A tangible testament to the robustness of these findings is the recyclability assessment, which resoundingly affirmed the enduring vigor of  $\text{Ti}_{16}\text{L}_4\text{-1}$ 's photocatalytic activity even across five successive cycles (Fig. 7D). It's worth highlighting that  $\text{Ti}_{16}\text{L}_4\text{-1}$  not only retained its remarkable photocatalytic stability but also withstood scrutiny through XPS, MALDI-TOF-MS and TEM morphology studies (Fig. S58–S60†). Subsequent SCXRD analysis of  $\text{Ti}_{16}\text{L}_4\text{-1}$  post photocatalysis revealed subtle changes in the coordination environment of potential catalytic sites (Ti2 and Ti3) within the cluster, which further substantiates the catalytic activity of these flexible coordination sites (Fig. S57†). Following the culmination of five cycles, the catalyst could be efficiently reclaimed from the reaction milieu. A analysis of the resultant supernatant *via* Inductively-Coupled Plasma (ICP) revealed an astonishingly low Ti leakage rate from  $\text{Ti}_{16}\text{L}_4\text{-1}$ , accounting for a mere 0.01% of the total Ti content. This revelation underscores not only the catalyst's enduring stability but also its marked potential for pragmatic, real-world applications. To unravel the origin of the carbon in the generated CO, an isotopic experiment leveraging  $^{13}\text{CO}_2$  as the carbon source was conducted. The gas chromatography-mass spectrometry (GC-MS) identified the production of  $^{13}\text{CO}$ , confirming the unequivocal derivation of CO from  $\text{CO}_2$  (Fig. 7E).

The investigation into electron transfer in the catalytic process unveiled the flat band potentials of  $\text{Ti}_{16}\text{L}_4\text{-1/2/3}$  *via* Mott-Schottky plots, pinpointing their calculated conduction band minimum (CBM) values. These CBM values resided at  $-0.86$ ,  $-0.91$ , and  $-1.08$  V vs. the normal hydrogen electrode (NHE), respectively. Strikingly, all LUMO potentials exhibited a notably more negative profile in contrast to  $\text{CO}_2/\text{CO}$  ( $-0.51$  V vs. NHE), signifying their inherent suitability for  $\text{CO}_2\text{RR}$  (Fig. 7F). Previous reports have indicated that the LUMO of  $[\text{Ru}(\text{bpy})_3]\text{Cl}_2\cdot 6\text{H}_2\text{O}$  is  $-1.27$  V vs. NHE.<sup>62</sup> Under irradiation,  $[\text{Ru}(\text{bpy})_3]\text{Cl}_2\cdot 6\text{H}_2\text{O}$  transits to an excited state, subsequently reductively quenched by TEOA, yielding a reducing photosensitizer. Given that the CBM values of  $\text{Ti}_{16}\text{L}_4\text{-1/2/3}$  lie below the LUMO of  $[\text{Ru}(\text{bpy})_3]\text{Cl}_2\cdot 6\text{H}_2\text{O}$ , the excited electrons emanating from the reduced photosensitizer migrate to  $\text{Ti}_{16}\text{L}_4$ , setting off the activation of adsorbed  $\text{CO}_2$  on its surface. This orchestrated sequence culminates in the reduction of  $\text{CO}_2$  to CO, subsequently liberating the product. We further substantiated the photocatalytic mechanism through electron paramagnetic resonance (EPR) experiments (Fig. S53†). The experimental findings reveal that in the absence of light irradiation within an  $\text{N}_2$  atmosphere, the reaction system



Fig. 8 *In situ* FTIR spectra on  $\text{Ti}_{16}\text{L}_4\text{-1}$ .

involving  $\text{Ti}_{16}\text{L}_4\text{-1}$  and the sacrificial agent did not exhibit any ESR signals. Nevertheless, when subjected to visible light irradiation, distinct  $\text{Ti}^{3+}$  signals were observed, corresponding to  $g$  value of 1.945. This observation implies that the photoexcited electrons transfers to  $\text{Ti}^{4+}$ , leading to their reduction to  $\text{Ti}^{3+}$ . Concurrently, TEOA serves as the sacrificial agent to neutralize the photogenerated holes. The intensity of the  $\text{Ti}^{3+}$  signal gradually increased with extended irradiation time. Upon exposure of the reaction system to a  $\text{CO}_2$  atmosphere, the ESR signal of  $\text{Ti}^{3+}$  diminished, signifying the involvement of photogenerated  $\text{Ti}^{3+}$  in  $\text{CO}_2\text{RR}$ . ESR results affirm that  $\text{Ti}^{4+}$  within  $\text{Ti}_{16}\text{L}_4\text{-1}$  function as the active centers for photocatalytic  $\text{CO}_2\text{RR}$ , providing further support for this mechanism.

To probe the  $\text{CO}_2$  radical and other reaction intermediates in the photocatalytic reaction, the  $\text{Ti}_{16}\text{L}_4\text{-1}$  is investigated by *in situ* diffuse reflectance infrared Fourier transform spectroscopy (Fig. 8). Under dark conditions post  $\text{CO}_2$  pretreatment,  $\text{Ti}_{16}\text{L}_4\text{-1}$  displayed prominent peaks around  $2348\text{ cm}^{-1}$ , which is associated with the asymmetric stretching of absorbed  $\text{CO}_2$ .<sup>59</sup> Exposure to a  $\text{CO}_2$  atmosphere under light for 10 minutes resulted in several new peaks: monodentate carbonate ( $\text{m-CO}_3^{2-}$ ) at 1351, 1451, and  $1508\text{ cm}^{-1}$ ; bidentate carbonate ( $\text{b-CO}_3^{2-}$ ) at 1290 and  $1543\text{ cm}^{-1}$ ; and bicarbonate ( $\text{HCO}_3^-$ ) at  $1406$  and  $1656\text{ cm}^{-1}$ . These carbonates and bicarbonates likely formed from  $\text{CO}_2$  reacting with water vapor. Notably, the  $\text{CO}_2^-$  signal at  $1713\text{ cm}^{-1}$  intensified with prolonged irradiation, indicating the formation of the  $\text{CO}_2$  radical, a key intermediate in  $\text{CO}_2$  photoreduction to  $^*\text{COOH}$ .<sup>52,53</sup> Furthermore,  $^*\text{COOH}$  groups, crucial in  $\text{CO}_2$  reduction to  $\text{CO}$ , were identified at  $1338$  and  $1584\text{ cm}^{-1}$ , with increasing peak intensities under light exposure, suggesting light-induced formation.<sup>63</sup> Additionally, absorption peaks for  $^*\text{CO}$  and gaseous  $\text{CO}$  at  $1708$  and  $2116\text{ cm}^{-1}$  respectively, provided further evidence of the reaction pathway.<sup>64</sup> Therefore, according to the above analysis, a rational  $\text{CO}_2$  photoreduction mechanism was proposed (Fig. S54†):  $\text{CO}_2$  was initially adsorbed on the  $\text{Ti}^{3+}$ . Subsequently, the adsorbed  $^*\text{CO}_2$  received electrons from  $\text{Ti}^{3+}$  and with protons to form the  $^*\text{COOH}$  during irradiation. Thereafter, the deprotonation of the  $^*\text{COOH}$  intermediate further generation of  $\text{CO}$ , and ultimately desorbs to obtain  $\text{CO}$  molecules.

## Conclusions

In summary, this research marks a pioneering exploration into the realm of  $\text{Ti}^{IV}$ -based metallamacrocycles, unveiling a host of

novel insights and underlining its significance in the broader landscape of scientific inquiry. Foremost, the unparalleled stability demonstrated by the  $\text{Ti}_{16}\text{L}_4$  stands out as a groundbreaking revelation. Its exceptional resilience against a spectrum of challenges, encompassing organic solvents, concentrated acids, and alkali, attests to a robustness that guarantees structural integrity and endurance across diverse environmental contexts. This stability not only solidifies the cluster's foundation but also paves the way for its application across a multitude of conditions. Equally groundbreaking is the abundance of coordination anchors inherent on the  $\text{Ti}_{16}\text{L}_4$ 's surface, effectively serving as a fertile ground for subsequent ligand adaptations. The dual pathways for facile ligand exchange form the bedrock of a versatile platform, facilitating the creation of microporous materials endowed with a plethora of surface functionalities. These microporous materials with coordinating sites on their surfaces are promising as carriers for loading noble metal nanoparticles, constructing heterogeneous catalysts with highly efficient catalytic activity. Another revelation lies in the crystallization potential of the macrocycle, giving rise to three distinct phases contingent upon the employed surface ligands. This revelation casts a revelatory spotlight on the predominant forces orchestrating the formation of diverse cluster-packing modes within the crystal lattice. The capability to manipulate surface functionalities and packing arrangements not only broadens our fundamental understanding but also unlocks a trove of opportunities for finely honing chemical and physical attributes to align with specific applications.

## Data availability

The data that support the findings of this study are available in the main text and the ESI.†

## Author contributions

C. L. supervised the project and conceived the idea. Y. Q. T. and L. F. D carried out synthesis, characterization and photocatalytic experiment of clusters. C. L. and Y. Q. T. wrote the manuscript. All authors discussed the experimental results.

## Conflicts of interest

There are no conflicts of interest to declare.

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## Notes and references

- 1 M. Dahl, Y. D. Liu and Y. D. Yin, *Chem. Rev.*, 2014, **114**, 9853–9889.
- 2 Y. Ma, X. L. Wang, Y. S. Jia, X. B. Chen, H. X. Han and C. Li, *Chem. Rev.*, 2014, **114**, 9987–10043.

3 X. Chen, L. Liu, P. Y. Yu and S. S. Mao, *Science*, 2011, **331**, 746–750.

4 P. Coppens, Y. Chen and E. Trzop, *Chem. Rev.*, 2014, **114**, 9645–9661.

5 L. Rozes and C. Sanchez, *Chem. Soc. Rev.*, 2011, **40**, 1006–1030.

6 P. D. Matthews, T. C. King and D. S. Wright, *Chem. Commun.*, 2014, **50**, 12815–12823.

7 J. B. Benedict, R. Freindorf, E. Trzop, J. Cogswell and P. Coppens, *J. Am. Chem. Soc.*, 2010, **132**, 13669–13671.

8 W. H. Fang, L. Zhang and J. Zhang, *Chem. Soc. Rev.*, 2018, **47**, 404–421.

9 Q. Y. Zhu and J. Dai, *Coord. Chem. Rev.*, 2021, **430**, 213664.

10 S. Yuan, J. S. Qin, H. Q. Xu, J. Su, D. Rossi, Y. P. Chen, L. L. Zhang, C. Lollar, Q. Wang, H. L. Jiang, D. H. Son, H. Y. Xu, Z. H. Huang, X. D. Zou and H. C. Zhou, *ACS Cent. Sci.*, 2018, **4**, 105–111.

11 J. D. Sokolow, E. Trzop, Y. Chen, J. Tang, L. J. Allen, R. H. Crabtree, J. B. Benedict and P. Coppens, *J. Am. Chem. Soc.*, 2012, **134**, 11695–11700.

12 G. Zhang, C. Liu, D.-L. Long, L. Cronin, C. H. Tung and Y. Wang, *J. Am. Chem. Soc.*, 2016, **138**, 11097–11100.

13 Y. Guo, J. L. Hou, W. Luo, Z. Q. Li, D. H. Zou, Q. Y. Zhu and J. Dai, *J. Mater. Chem. A*, 2017, **5**, 18270–18275.

14 D. H. Zou, L. N. Cui, P. Y. Liu, S. Yang, Q. Y. Zhu and J. Dai, *Inorg. Chem.*, 2019, **58**, 9246–9252.

15 Y. Lv, J. Cheng, A. Steiner, L. Gan and D. S. Wright, *Angew. Chem., Int. Ed.*, 2014, **53**, 1934–1938.

16 L. Zhang, X. Fan, X. F. Yi, X. Lin and J. Zhang, *Acc. Chem. Res.*, 2022, **55**(21), 3150–3161.

17 L. Geng, C. H. Liu, S. T. Wang, W.-H. Fang and J. Zhang, *Angew. Chem., Int. Ed.*, 2021, **133**, 21596–21603.

18 M.-Y. Gao, F. Wang, Z. G. Gu, D. X. Zhang, L. Zhang and J. Zhang, *J. Am. Chem. Soc.*, 2016, **138**, 2556–2559.

19 X. Fan, J. Wang, K. Wu, L. Zhang and J. Zhang, *Angew. Chem., Int. Ed.*, 2019, **58**, 1320–1323.

20 W. H. Fang, L. Zhang and J. Zhang, *J. Am. Chem. Soc.*, 2016, **138**, 7480–7483.

21 G. Zhang, W. Li, C. Liu, J. Jia, C. H. Tung and Y. Wang, *J. Am. Chem. Soc.*, 2018, **140**, 66–69.

22 J. X. Liu, M. Y. Gao, W. H. Fang, L. Zhang and J. Zhang, *Angew. Chem., Int. Ed.*, 2016, **55**, 5160–5165.

23 L. Geng, C. H. Liu, S. T. Wang, W. H. Fang and J. Zhang, *Angew. Chem., Int. Ed.*, 2020, **59**, 16735–16740.

24 Y. Q. Tian, Y. S. Cui, W. D. Yu, C. Q. Xu, X. Y. Yi, J. Yan, J. Li and C. Liu, *Chem. Commun.*, 2022, **52**, 6028–6031.

25 Y. Q. Tian, Y. S. Cui, J. H. Zhu, C. Q. Xu, X. Y. Yi, J. Li and C. Liu, *Chem. Commun.*, 2022, **58**, 9034–9037.

26 S. Y. Zhou, C. P. Li, H. Fu, J. Cao, J. Zhang and L. Zhang, *Chem.–Eur. J.*, 2020, **26**, 6894–6898.

27 M. Liu, W. P. Liao, C. Hu, S. C. Du and H. J. Zhang, *Angew. Chem., Int. Ed.*, 2012, **51**, 1585–1588.

28 W. Y. Pei, G. Xu, J. Yang, H. Wu, B. Chen, W. Zhou and J. F. Ma, *J. Am. Chem. Soc.*, 2017, **139**, 7648–7656.

29 H. T. Han, L. Kan, P. Li, G. S. Zhang, K. Y. Li, W. P. Liao, Y. L. Liu, W. Chen and C. H. T. Hu, *Sci. China: Chem.*, 2021, **64**, 426–431.

30 X. X. Hang, B. Liu, X. F. Zhu, S. T. Wang, H. T. Han, W. P. Liao, Y. L. Liu and C. H. Hu, *J. Am. Chem. Soc.*, 2016, **138**, 2969–2972.

31 S. Wang, X. Gao, X. Hang, X. Zhu, H. Han, X. Li, W. Liao and W. Chen, *J. Am. Chem. Soc.*, 2018, **140**, 6271–6277.

32 Z. Wang, H. F. Su, Y. W. Gong, Q. P. Qu, Y. F. Bi, C. H. Tung, D. Sun and L. S. Zheng, *Nat. Commun.*, 2020, **11**, 308.

33 B. Hou, H. Y. Zheng, K. H. Zhang, Q. Wu, C. Qin, C. Y. Sun, Q. H. Pan, Z. H. Kang, X. L. Wang and Z. M. Su, *Chem. Sci.*, 2023, **14**, 8962–8969.

34 A. J. Tasiopoulos, A. Vinslava, W. Wernsdorfer, K. A. Abboud and G. Christou, *Angew. Chem., Int. Ed.*, 2004, **43**, 2117–2121.

35 H. Kitagawa, H. Ohtsu and M. Kawano, *Angew. Chem., Int. Ed.*, 2013, **52**, 12395–12399; *Angew. Chem.*, 2013, **125**, 12621–12625.

36 Z. M. Zhang, Y. G. Li, S. Yao, E. B. Wang, Y. H. Wang and R. Clerac, *Angew. Chem., Int. Ed.*, 2009, **48**, 1581–1584.

37 R. A. Scullion, A. J. Surman, F. Xu, J. S. Mathieson, D. L. Long, F. Haso, T. Liu and L. Cronin, *Angew. Chem., Int. Ed.*, 2014, **53**, 10032–10037.

38 C. Zhao, Y.-Z. Han, S. Dai, X. Chen, J. Yan, W. Zhang, H. Su, S. Lin, Z. Tang, B. K. Teo and N. Zheng, *Angew. Chem., Int. Ed.*, 2017, **56**, 16252–16256.

39 S. Øien-Ødegaard, C. Bazioti, E. A. Redekop, O. Prytz, K. P. Lillerud and U. Olsbye, *Angew. Chem., Int. Ed.*, 2020, **59**, 21397–21402.

40 X. Y. Zheng, Y. H. Jiang, G. L. Zhuang, D. P. Liu, H. G. Liao, X. J. Kong, L. S. Long and L. S. Zheng, *J. Am. Chem. Soc.*, 2017, **139**, 18178–18181.

41 Y. Q. Tian, L. F. Dai, W. L. Mu, W. D. Yu, J. Yan and C. Liu, CCDC 2174429: *Experimental Crystal Structure Determination*, DOI: [10.5517/cedc.csd.cc2bznww](https://doi.org/10.5517/cedc.csd.cc2bznww).

42 Y. Q. Tian, L. F. Dai, W. L. Mu, W. D. Yu, J. Yan and C. Liu, CCDC 2174430: *Experimental Crystal Structure Determination*, DOI: [10.5517/cedc.csd.cc2bznxx](https://doi.org/10.5517/cedc.csd.cc2bznxx).

43 Y. Q. Tian, L. F. Dai, W. L. Mu, W. D. Yu, J. Yan and C. Liu, CCDC 2174431: *Experimental Crystal Structure Determination*, DOI: [10.5517/cedc.csd.cc2bzny](https://doi.org/10.5517/cedc.csd.cc2bzny).

44 Y. Q. Tian, L. F. Dai, W. L. Mu, W. D. Yu, J. Yan and C. Liu, CCDC 2174432: *Experimental Crystal Structure Determination*, DOI: [10.5517/cedc.csd.cc2bzpz](https://doi.org/10.5517/cedc.csd.cc2bzpz).

45 Y. Q. Tian, L. F. Dai, W. L. Mu, W. D. Yu, J. Yan and C. Liu, CCDC 2174433: *Experimental Crystal Structure Determination*, DOI: [10.5517/cedc.csd.cc2bzp01](https://doi.org/10.5517/cedc.csd.cc2bzp01).

46 Y. Q. Tian, L. F. Dai, W. L. Mu, W. D. Yu, J. Yan and C. Liu, CCDC 2174435: *Experimental Crystal Structure Determination*, DOI: [10.5517/cedc.csd.cc2bzp23](https://doi.org/10.5517/cedc.csd.cc2bzp23).

47 Y. Q. Tian, L. F. Dai, W. L. Mu, W. D. Yu, J. Yan and C. Liu, CCDC 2174434: *Experimental Crystal Structure Determination*, DOI: [10.5517/cedc.csd.cc2bzp12](https://doi.org/10.5517/cedc.csd.cc2bzp12).

48 Y. Q. Tian, L. F. Dai, W. L. Mu, W. D. Yu, J. Yan and C. Liu, CCDC 2174437: *Experimental Crystal Structure Determination*, DOI: [10.5517/cedc.csd.cc2bzp45](https://doi.org/10.5517/cedc.csd.cc2bzp45).

49 Y. Q. Tian, L. F. Dai, W. L. Mu, W. D. Yu, J. Yan and C. Liu, CCDC 2174438: *Experimental Crystal Structure Determination*, DOI: [10.5517/cedc.csd.cc2bzp56](https://doi.org/10.5517/cedc.csd.cc2bzp56).



50 J. Zhou, J. Li, L. Kan, L. Zhang, Q. Huang, Y. Yan, Y. Chen, J. Liu, S. L. Li and Y. Q. Lan, *Nat. Commun.*, 2022, **13**, 4681.

51 N. Li, J.-J. Liu, J.-W. Sun, B.-X. Dong, L.-Z. Dong, S.-J. Yao, Z. F. Xin, S.-L. Li and Y.-Q. Lan, *Green Chem.*, 2020, **22**, 5325–5332.

52 N. Li, J.-M. Lin, R.-H. Li, J.-W. Shi, L.-Z. Dong, J. Liu, J. He and Y.-Q. Lan, *J. Am. Chem. Soc.*, 2023, **145**(29), 16098–16108.

53 J.-J. Liu, N. Li, J.-W. Sun, J. Liu, L.-Z. Dong, S.-J. Yao, L. Zhang, Z.-F. Xin, J.-W. Shi, J.-X. Wang, S.-L. Li and Y.-Q. Lan, *ACS Catal.*, 2021, **11**, 4510–4519.

54 N. Li, J. Liu, J. J. Liu, L. Z. Dong, S. L. Li, B. X. Dong, Y. H. Kan and Y. Q. Lan, *Angew. Chem., Int. Ed.*, 2019, **58**, 17260–17264.

55 Y. Fu, D. Sun, Y. Chen, R. Huang, Z. Ding, X. Fu and Z. Li, *Angew. Chem., Int. Ed.*, 2012, **51**, 3364–3367.

56 R. Li, J. H. Hu, M. S. Deng, H. L. Wang, X. J. Wang, Y. L. Hu, H. L. Jiang, J. Jiang, Q. Zhang, Y. Xie and Y. J. Xiong, *Adv. Mater.*, 2014, **26**, 4783–4788.

57 S. B. Wang, W. S. Yao, J. L. Lin, Z. X. Ding and X. C. Wang, *Angew. Chem., Int. Ed.*, 2014, **53**, 1034–1308.

58 J. Zhao, Q. Wang, C. Y. Sun, T. T. Zheng, L. K. Yan, M. T. Li, K. Z. Shao, X. L. Wang and Z. M. Su, *J. Mater. Chem. A*, 2017, **5**, 12498.

59 H.-X. Zhang, Q.-L. Hong, J. Li, F. Wang, X. S. Huang, S. M. Chen, W. G. Tu, D. S. Yu, R. Xu, T. H. Zhou and J. Zhang, *Angew. Chem., Int. Ed.*, 2019, **58**, 11752–11756.

60 X. X. Li, L. Zhang, L. Yuan, T. Wang, L. Z. Dong, K. Huang, J. Liu and Y. Q. Lan, *Chem. Eng. J.*, 2022, **442**, 136157.

61 J. Du, Y. Ma, X. Xin, H. Na, Y. Zhao, H. Tan, Z. Han, Y. Li and Z. Kang, *Chem. Eng. J.*, 2020, **398**, 125518.

62 P. P. Bag, X. S. Wang, P. Sahoo, J. H. Xiong and R. Cao, *Catal. Sci. Technol.*, 2017, **7**, 5113–5119.

63 G. Fan, R. X. Li, S. Z. Yang, X. Y. Zhang, H. Y. Jian, J. J. Urban and W.-Y. Sun, *Angew. Chem., Int. Ed.*, 2023, **62**, e202216232.

64 J.-P. Dong, Y. Xu, X.-G. Zhang, H. Zhang, L. Yao, R. Wang and S.-Q. Zang, *Angew. Chem., Int. Ed.*, 2023, e202313648.

