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Halogen bonding with carbon: directional assembly of non-derivatised aromatic carbon systems into robust supramolecular ladder architectures†

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Carbon, although the central element in organic chemistry, has been traditionally neglected as a target for directional supramolecular interactions. The design of supramolecular structures involving carbon-rich molecules, such as arene hydrocarbons, has been limited almost exclusively to non-directional π -stacking, or derivatisation with heteroatoms to introduce molecular assembly recognition sites. As a result, the predictable assembly of non-derivatised, carbon-only π -systems using directional non-covalent interactions remains an unsolved fundamental challenge of solid-state supramolecular chemistry. Here, we propose and validate a different paradigm for the reliable assembly of carbon-only aromatic systems into predictable supramolecular architectures: not through non-directional π -stacking, but *via* specific and directional halogen bonding. We present a systematic experimental, theoretical and database study of halogen bonds to carbon-only π -systems ($C-I \cdots \pi_C$ bonds), focusing on the synthesis and structural analysis of cocrystals with diversely-sized and -shaped non-derivatised arenes, from one-ring (benzene) to 15-ring (dicyrnylene) polycyclic aromatic hydrocarbons (PAHs), and fullerene C_{60} , along with theoretical calculations and a systematic analysis of the Cambridge Structural Database. This study establishes $C-I \cdots \pi_C$ bonds as directional interactions to arrange planar and curved carbon-only aromatic systems into predictable supramolecular motifs. In >90% of herein presented structures, the $C-I \cdots \pi_C$ bonds to PAHs lead to a general ladder motif, in which the arenes act as the rungs and halogen bond donors as the rails, establishing a unique example of a supramolecular synthon based on carbon-only molecules. Besides fundamental importance in the solid-state and supramolecular chemistry of arenes, this synthon enables access to materials with exciting properties based on simple, non-derivatised aromatic systems, as seen from large red and blue shifts in solid-state luminescence and room-temperature phosphorescence upon cocrystallisation.

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Introduction

Polycyclic aromatic hydrocarbons (PAHs) are an extensive class of carbon-based molecules, estimated to hold 10–15% of all

carbon in the known universe.^{1–3} PAH-based molecules are ubiquitous in the design of organic electronics^{4,5} and organic light emitting diodes (OLEDs).^{6,7} While the arrangement of PAH units in solids is of critical importance for their optical and electronic properties,⁸ controlling the assembly of PAHs in crystalline materials is a persistent challenge of organic solid-state chemistry.⁹ In most cases, guiding the arrangement of PAH units in the solid state requires derivatisation, either to introduce sterically demanding groups that modify molecular packing,¹⁰ or to introduce recognition sites for the formation of multi-component crystals (cocrystals) by directional interactions such as hydrogen (HB)^{11–13} or halogen bonding (XB).¹⁴

Such derivatisation strategies are, however, not applicable to the assembly of pristine, non-derivatised PAHs. Indeed, while carbon is central to organic chemistry, it is rarely considered a target of supramolecular recognition. Directional interactions such as HB ¹⁵ and XB ^{16–18} are generally regarded as insufficiently

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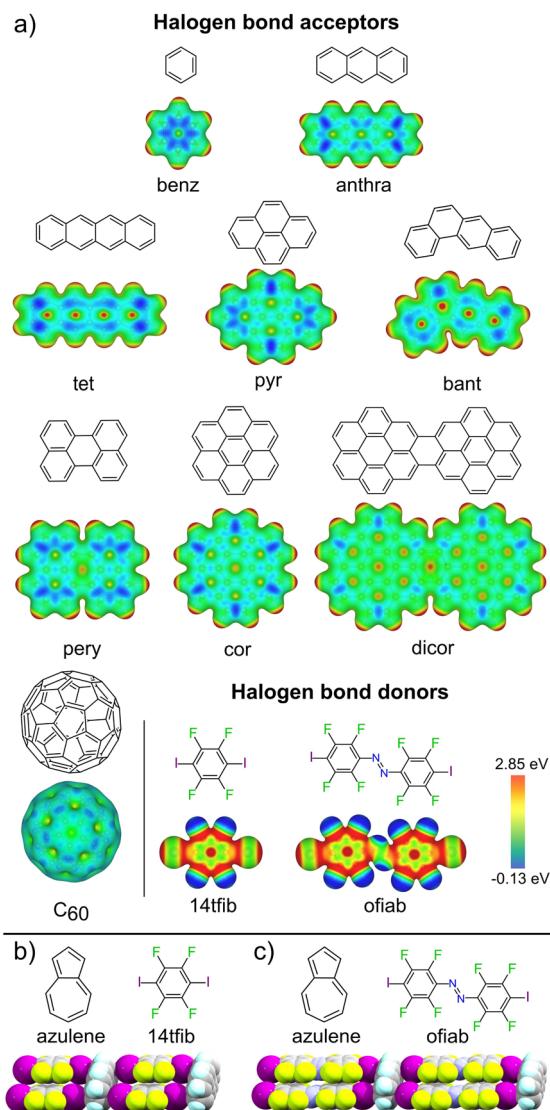


Fig. 1 (a) Halogen bond acceptor and donor molecules explored in this work, along with corresponding electrostatic surface potential (ESP) maps, with isosurfaces plotted at 0.01 a.u. Fragments of the halogen-bonded $\text{C}-\text{I}\cdots\pi_{\text{C}}$ supramolecular ladder seen in the cocrystals of: (b) (azulene)(14tfib)₂ and (c) (azulene)(ofiab)₂ along with corresponding molecular diagrams.⁴³

robust to enable predictable, directional assembly of carbon-only π -systems. As a result, approaches to molecular recognition, and in particular to cocrystallisation, of PAHs have traditionally been limited to non-directional face-to-face $\pi\cdots\pi$ stacking.¹⁹ This presents the design of supramolecular architectures based on directional assembly of carbon-only systems as an unsolved fundamental problem of solid-state supramolecular chemistry, and of the supramolecular chemistry of carbon in general.²⁰⁻²²

Halogen bonding has emerged as a versatile directional interaction in the crystalline solid state, offering access to a wider range of acceptor atoms than seen for hydrogen bonding.²³ Scattered structural reports show that halogen bonds can form between electron-deficient iodine atoms as

donors and flat, electron-rich π -systems, such as anilines, as acceptors.²⁴⁻³⁴ While cocrystals exhibiting individual XBs to carbon have been reported or theoretically studied, the possibility of using such interactions for the predictable assembly of supramolecular architectures based on PAHs has not been established.³⁵⁻⁴² Recently,⁴³ we observed that cocrystals of azulene with XB donors 1,4-diiodotetrafluorobenzene (**tfib**) or *trans*-octafluoro-4,4'-diiodoazobenzene (**ofiab**) (Fig. 1b and c) exhibit a ladder-like halogen-bonded motif involving aromatic carbon, in which the PAHs act as the rungs and the XB donors as rails, identical to the motif seen in a cocrystal of naphthalene³⁰ with **14tfib**. This unexpected observation of the same $\text{C}-\text{I}\cdots\pi_{\text{C}}$ ladder-like motif in three different cocrystals of naphthalene and azulene suggested an unexpected role for XB as a unique, possibly general tool for directional assembly of non-derivatised PAHs – contrasting the traditionally relied upon non-directional π -stacking.

We now provide an extensive and systematic study that demonstrates halogen bonding as a long-overlooked, reliable tool for the directional assembly of non-derivatised carbon-only aromatic systems. Our study, addressing diversely-shaped and -sized PAHs, as well as fullerene C_{60} (Fig. 1a), reveals the persistent formation of cocrystals based on a highly reproducible⁴⁴⁻⁴⁶ supramolecular ladder-like motif (Fig. 1b, c, and 2) of $\text{C}-\text{I}\cdots\pi_{\text{C}}$ halogen bonds. The robustness of this motif presents it as a unique example of a supramolecular synthon⁴⁷ for the assembly of exclusively carbon-based aromatic systems through directional interactions, independent of molecular size, shape, or presence of $\pi\cdots\pi$ or $\text{C}-\text{H}\cdots\pi$ contacts. The presented set of cocrystals involving different combinations of 9 XB acceptors and 2 donors, along with theoretical calculations and a systematic analysis of the Cambridge Structural Database (CSD),⁴⁸ establishes a different paradigm for the crystal engineering based on aromatic carbon: not *via* surface-based π -stacking, but by directional halogen bonding. The potential of this approach is evident by luminescence studies of select cocrystals based on pyrene, coronene and perylene, demonstrating extensive modification of emission properties due to halogen bond-directed assembly.

Results and discussion

Linear arenes (acenes)

Our first target was cocrystallisation of **14tfib** with benzene (**benz**), the smallest possible arene that could be addressed in this study. Dissolving **14tfib** in **benz** followed by slow evaporation produced large, colorless plate-like crystals. Removal of the crystals from the crystallisation vessel at room temperature, however, resulted in immediate degradation, evident by a loss of crystal transparency. Cooling the sample in dry ice enabled successful handling of the crystal and collection of X-ray single crystal diffraction data at 180 K (see ESI, Table S1†). Structure determination revealed that the crystals exhibit the composition (**benz**)(**14tfib**), based on one-dimensional (1D) supramolecular chains of alternating **benz** and **14tfib** molecules, held by $\text{C}-\text{I}\cdots\pi_{\text{C}}$ halogen bonds between **14tfib** iodine atoms and each face of **benz** molecules (Fig. 2a). Specifically, each **14tfib**



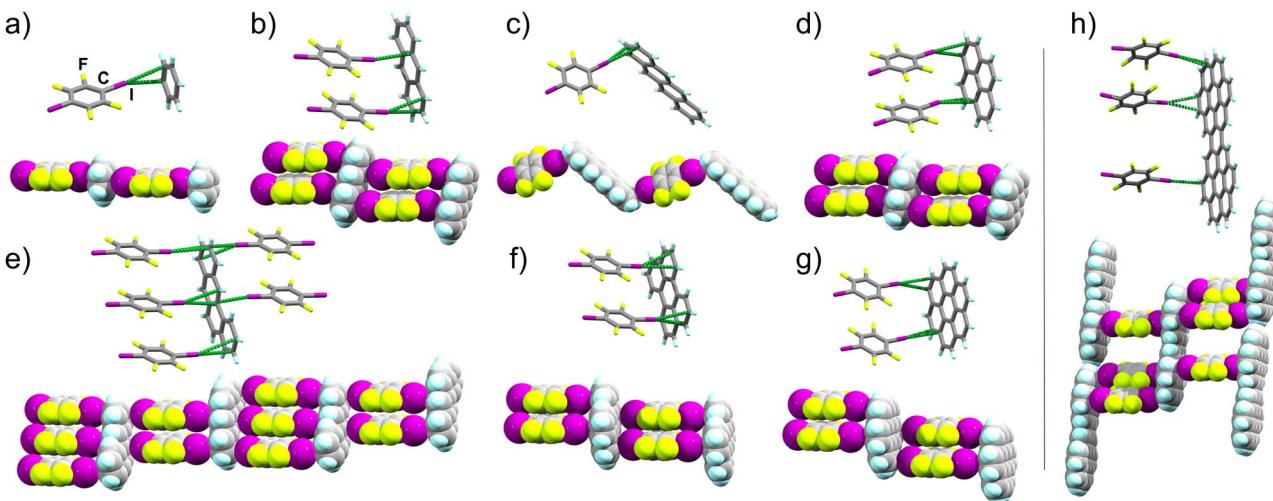


Fig. 2 Stick model representation of halogen-bonded fragments in explored **14tfib** cocrystals, with extended motifs shown using space-filling models: (a) **(benz)(14tfib)** (CCDC code 2240199), (b) **(anthra)(14tfib)₂** (CCDC code 2249511, also CSD JEJXOP),⁵⁵ (c) **(tet)(14tfib)** (CCDC code 2240203), (d) **(pyr)(14tfib)₂** (CCDC code 2240200), (e) **(bant)₂(14tfib)₅** (CCDC code 2281269), (f) **(pery)(14tfib)₂** (CCDC code 2240202), (g) **(cor)(14tfib)₂** (CCDC code 2240204), (h) **(dicor)(14tfib)₃** (CCDC code 2240207). The I...C_π intermolecular distances shorter than the sum of van der Waals radii for iodine and carbon atoms (3.68 Å) are shown as green dotted lines.⁴⁷

molecule forms multiple short I...C_π halogen-bonding contacts to the adjacent **benz** molecules, with intermolecular I...C distances ($d_{I...C}$) of 3.48(2) Å and 3.51(2) Å, with the corresponding C-I...C angles ($∠_{C-I...C}$) of 165° and 166°, respectively (Table 1, also ESI Table S2.†). The $d_{I...C}$ separations are shorter than the expected sum of van der Waals radii (3.68 Å) for carbon and iodine.⁴⁹

The simplicity of components makes it surprising that **(benz)(14tfib)** was not reported previously. The formation of halogen-bonded cocrystals with **benz**, however, is not without precedent, as 1D motifs similar to that in **(benz)(14tfib)** have been found with Br₂ (ref. 50) and 1,4-diiodotetrachlorobenzene as XB donors.⁵¹ While the $d_{I...C}$ distances in **(benz)(14tfib)** are ~5% shorter than the sum of corresponding van der Waals radii, the shortest distance between the iodine atom of a **14tfib** donor and a **benz** acceptor molecule is to the centroid of a carbon–carbon bond (3.42(1) Å). This is consistent with previously noted optimal geometry for halogen bonding to π-systems,^{36,37} and with the calculated electrostatic surface potential (ESP) surface of **benz** (Fig. 1a), showing that the areas of most negative ESP, expected to be beneficial for halogen bonding, are located on the arene π-bonds.

Bulk analysis of **(benz)(14tfib)** was hindered by rapid decomposition in air. Eventually, the formation of bulk cocrystal in a slurry of **14tfib** in excess **benz** was confirmed by powder X-ray diffraction (PXRD) analysis of a sample under a plastic-wrap cover (see ESI, Fig. S1†).⁵² The **(benz)(14tfib)** structure confirms that the simplest aromatic hydrocarbon can be used as an acceptor in forming C-I...C_π chains (Fig. 2a) reminiscent of those in cocrystals of naphthalene and azulene with **14tfib** or **ofiab**.^{30,43}

Next, we targeted anthracene (**anthra**) and tetracene (**tet**) as XB acceptors. For **anthra**, a cocrystal of composition

(anthra)(14tfib)₂ was obtained by mechanochemical liquid-assisted grinding (LAG),^{53,54} using nitromethane as a liquid additive (ESI, Fig. S2†).⁵⁵ Diffraction-quality single crystals of **(anthra)(14tfib)₂** were obtained by dissolving the mechanochemically made material in hot CH₂Cl₂, rapidly cooling in an ice bath for *ca.* 10 seconds, followed by slow evaporation. Structure analysis revealed that **(anthra)(14tfib)₂** is based on the ladder-like motif (Fig. 2b) with each face of the arene forming two C-I...C_π bonds with **14tfib** molecules. The $d_{I...C}$ distances ranged from 3.427(9) Å to 3.583(7) Å, with $∠_{C-I...C}$ angles in the range 157° to 173°. The distance between the iodine atom and the centroid of the relevant carbon–carbon bond was even shorter, at 3.366(5) Å, and no other short contacts were observed that would indicate C–H...π hydrogen bonding or π...π stacking between **anthra** molecules. The structure of **(anthra)(14tfib)₂** was also recently reported by Azzali *et al.* and our experimental data fully agrees with it.⁵⁵

In the case of **tet**, the cocrystal **(tet)(14tfib)** was obtained in the form of diffraction-quality single crystals by slow evaporation of a solution of **tet** and **14tfib** in 1,2,4-trichlorobenzene. Unlike the cocrystal containing **anthra**, the **(tet)(14tfib)** cocrystal does not exhibit the ladder-like motif, but instead presents a zig-zag architecture of alternating **tet** and **14tfib** molecules interconnected *via* short, linear C-I...C_π contacts (Fig. 2c, Table 1 and ESI Table S2†). The individual I...C distances are again shorter than the sum of van der Waals radii for C and I atoms, at 3.410(6) Å and 3.509(7) Å, with respective $∠_{C-I...C}$ angles of 163° and 174°. The corresponding distance of the **14tfib** iodine atom to the centroid of the relevant C–C bond on **tet** is again even shorter, at 3.387(5) Å, supporting the targeting of the π-system on the PAH. While **tet** is the only PAH in our study that did not produce the expected supramolecular ladder architecture, the



(tet)(14tfib) structure confirms the general reliability of C–I \cdots π_C halogen bonds for PAH cocrystallisation.

Non-linear arenes

In order to explore the possibility of using XB for the assembly of arenes beyond the acene family, we targeted cocrystallisation with pyrene (**pyr**), benzanthracene (**bant**), perylene (**pery**) and coronene (**cor**), constituted of four, five and six fused ring systems. Mechanochemical screening in all cases produced new cocrystal phases, with compositions (**pyr**)(14tfib)₂, (**bant**)₂(14tfib)₅, (**pery**)(14tfib)₂ and (**cor**)(14tfib)₂ (Fig. 2d–g). All four cocrystals were obtained in the form of diffraction-quality single crystals by crystallisation from CH₂Cl₂, and structural analysis revealed in each case the ladder motif analogous to the one seen with naphthalene,³⁰ azulene,⁴³ and **anthra**, with each face of the PAH participating in at least two linear C–I \cdots π_C halogen bonds. The halogen-bonded distances were in each cocrystal comparable, in the range from 3.4 Å to 3.6 Å, with \angle_{C-I-C} angles in the range from 156° to 174° (see Table 1 and ESI Table S2†). Each cocrystal was also obtained mechanochemically as a bulk powder, exhibiting PXRD patterns consistent with herein determined crystal structures (see ESI†). The stability and composition of each cocrystal was validated by differential scanning calorimetry and thermogravimetric analysis (DSC and TGA, respectively, see ESI†). The DSC analysis revealed that the cocrystals were generally stable up to *ca.* 110 °C, and upon decomposition exhibited a weight loss step consistent with a loss of **14tfib**. Structures of (**pyr**)(14tfib)₂, (**bant**)₂(14tfib)₅, (**pery**)(14tfib)₂, (**cor**)(14tfib)₂ and (**dicor**)(14tfib)₃ (Fig. 2d–h) further highlight the robustness of the supramolecular ladder motif based on C–I \cdots π_C XB interactions, even with expanded aromatic systems as XB acceptors.

In the case of **pyr**, two other cocrystals with **14tfib** have previously been reported, with different compositions: (**pyr**)(14tfib)²⁹ and (**pyr**)₄(14tfib).⁵⁶ Both of these previously reported stoichiomorphs contain **14tfib** molecules involved in either extended (for (**pyr**)(14tfib) cocrystal) or discrete (for (**pyr**)₄(14tfib) cocrystal) arene-perfluoroarene π -stacking motifs with **pyr** molecules. The **14tfib** molecules in these two prior cocrystals also form C–I \cdots C halogen bonds of *ca.* 3.5 Å to neighboring **pyr** units. It is, therefore, notable, that the (**pyr**)(14tfib)₂ phase reported here presents C–I \cdots π_C halogen bonding as the dominant intermolecular interaction, without any notable herringbone C–H \cdots π or arene-perfluoroarene π -stacking interactions seen in other stoichiomorphs (Fig. 2d, also see ESI Fig. S4†).^{29,56}

Cocrystallisation of **bant**, an isomer of **pyr** and **tet**, with **14tfib** resulted in a halogen-bonded cocrystal of composition (**bant**)₂(14tfib)₅. The cocrystal was found to again contain the anticipated ladder motif (Fig. 2e). The ladder motif in (**bant**)₂(14tfib)₅ is unique, however, with different faces of each **bant** unit alternatively taking part in either two or three C–I \cdots π_C halogen bonds.

Crystal structures of cocrystals (**pery**)(14tfib)₂ and (**cor**)(14tfib)₂ (Fig. 2f and g) are, to the best of our knowledge, the first examples of halogen bonding being used to form

cocrystals with large, non-substituted PAHs comprising >4 aromatic rings. To further examine the ability of C–I \cdots π_C bonds to organise large arenes, we focused on dicoronylene (**dicor**), comprised of 15 aromatic rings (Fig. 1a). A sample of **dicor** was synthesised according to the reported procedure,⁵⁷ by reaction of **cor** at 170 °C in a 3 : 1 by weight melt of AlCl₃ and NaCl, followed by sublimation (see ESI†). Attempts to obtain cocrystals of **14tfib** and **dicor** from common organic solvents were not successful due to poor arene solubility, with crystallisation eventually achieved from molten **14tfib** at 110 °C. Single crystal X-ray analysis of the ruby-red crystals revealed a cocrystal structure with composition (**dicor**)(14tfib)₃, exhibiting the C–I \cdots π_C ladder motif, but with each face of the arene now engaged in XBs with 3 separate **14tfib** donors (Fig. 2h). The XB distances lie in the range from 3.424(7) Å to 3.560(9) Å, with corresponding distances to the centroids of carbon–carbon bonds being in the range 3.327(6) Å to 3.579(8) Å. The halogen bonds exhibit a linear geometry, with \angle_{C-I-C} angles adopting values from 159–178°. Notably, the C–I \cdots π_C bonds form on the edges of each **dicor** molecule, consistent with the calculated arene ESP surface, which shows that the highest negative potential is localised on the molecular rim. This means that a large fraction of the **dicor** molecule surface is not likely to engage in halogen bonding, and could be accessible for other types of interactions. Indeed, the **dicor** molecules are also engaged in π – π stacking between adjacent halogen-bonded ladders.

Of the three symmetrically distinct **14tfib** molecules in the (**dicor**)(14tfib)₃ structure, two also participate in F \cdots F and C–H \cdots F contacts⁵⁸ with neighboring **14tfib** and **dicor** molecules, while one **14tfib** molecule also participates in F \cdots I contacts.

The formation of this (**dicor**)(14tfib)₃ shows that C–I \cdots π_C halogen bonds are sufficiently robust to enable the cocrystallisation of large PAHs – to the best of our knowledge, no cocrystal of a non-derivatised PAH of similar size to **dicor** has ever previously been reported. Attempts to produce a bulk sample of (**dicor**)(14tfib)₃ have so far been unsuccessful.

Non-planar aromatic systems: C₆₀

Next, we targeted a non-planar aromatic acceptor, buckminsterfullerene (C₆₀). While previous work reported that C₆₀ and **14tfib** do not form a cocrystal,³⁹ we found that milling of the two components produces a cocrystal of composition (C₆₀)(14tfib)₂. Dark-red block-shaped crystals of (C₆₀)(14tfib)₂ were subsequently obtained by slow evaporation of a toluene solution of the mechanochemically made material. Crystal structure analysis revealed that (C₆₀)(14tfib)₂ is based on interpenetrated square-grid topology (*sql*) nets of C–I \cdots π_C bonds (Fig. 3a) with d_{I-C} separation of 3.54(1) Å and halogen bond angle of 163°. The *sql*-nets are formed by C₆₀ molecules acting as four-fold nodes with **14tfib** molecules as framework linkers (Fig. 3b and c). The structure also exhibits short C \cdots F contacts between **14tfib** and C₆₀ molecules in adjacent nets (d_{F-C} = 3.17(1) Å; \angle_{C-F-C} = 102.8(4)°). Notably, nearest-neighbor C₆₀ molecules do not exhibit any short mutual contacts, and at room temperature do not exhibit the rotational disorder commonly seen in C₆₀-containing crystal structures.^{59–62} Analysis of



$(C_{60})(14tfib)_2$ by DSC and TGA reveals decomposition around 110 °C, evidenced by a single endothermic event in the DSC thermogram, accompanied by the loss of 50% of sample weight, consistent with the theoretical content of **14tfib** (52.7%).

Replacement of the halogen bond donor

In order to further explore the generality of the supramolecular ladder motif formed with PAHs, we next attempted cocrystallisation of **anthra** and **pyr** with **ofiab**, a longer XB donor. In each case, milling of the XB donor and acceptor followed by PXRD analysis revealed a new cocrystal phase, with compositions **(anthra)(ofiab)₂** and **(pyr)(ofiab)₂**. Crystallographic analysis of single crystals grown by recrystallisation of the mechanochemically-made material from CH_2Cl_2 revealed the

expected C–I··· π_C ladder-like architectures in both cases (Fig. 3d, e, Table 1 and ESI Table S2†). Importantly, the measured I···C and I···centroid distances are not appreciably different from those in **(anthra)(14tfib)₂** and **(pyr)(14tfib)₂**. Both **(anthra)(ofiab)₂** and **(pyr)(ofiab)₂** were found to melt around 160–170 °C, as established by DSC, with TGA showing that samples completely evaporate, in a single step, at temperatures around 200 °C.

Periodic DFT calculations

The formation of herein reported cocrystals was also investigated by plane-wave periodic-density functional theory (DFT) calculations in CASTEP,⁶³ using PBE⁶⁴ functionals combined with Grimme D3 semiempirical dispersion correction.⁶⁵

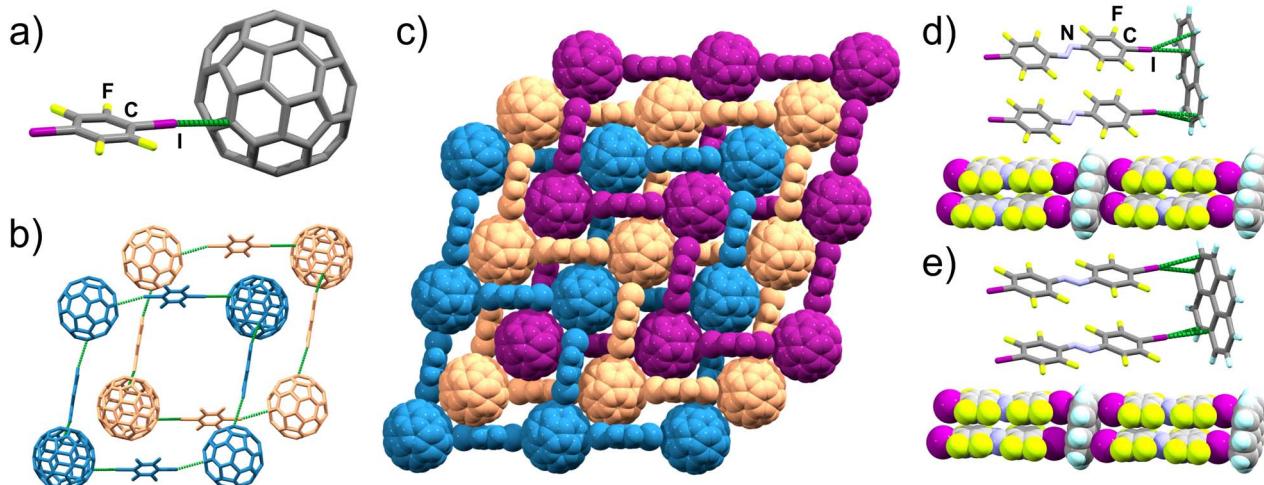


Fig. 3 Structural motifs in $(C_{60})(14tfib)_2$ and cocrystals of **anthra** and **pyr** with **ofiab**: (a) the C–I···C contact between iodine and C_{60} in $(C_{60})(14tfib)_2$ (CCDC code 2240201); (b) interpenetration of two sql-grids in $(C_{60})(14tfib)_2$; (c) space-filling representation of three distinct $(C_{60})(14tfib)_2$ sql-grids; (d) fragment of the supramolecular ladder in **(anthra)(ofiab)₂** (CCDC code 2240205) and (e) in **(pyr)(ofiab)₂** (CCDC code 2240206).

Table 1 Calculated energies of cocrystal formation (E_{calc} , in kJ mol^{-1}) from starting materials, halogen bond interaction energies (E_{XB} , in kJ mol^{-1}), and experimental C–I···C (in Å) distances for herein reported cocrystals.^a For structures where an arene molecules forms C–I··· π_C halogen bonds with multiple XB donors, the average E_{XB} is provided, with individual E_{XB} values in brackets

| Cocrystal | E_{calc} / kJ mol^{-1} | E_{XB} / kJ mol^{-1} | $d(\text{C–I} \cdots \text{C}_\pi)/\text{\AA}$ |
|---|--|---|---|
| (benz)(14tfib) | −10.4 ^b | −16.43 | 3.48(2), 3.51(2) |
| (anthra)(14tfib)₂ | −5.89 | −18.82 (−19.28, −18.36) | 3.427(9), 3.445(8), 3.583(7) |
| (tet)(14tfib) | −2.27 | −13.76 | 3.410(6), 3.509(7) |
| (pyr)(14tfib)₂ | −5.90 | −18.86 (−18.63, −19.10) | 3.515(8), 3.56(1), 3.50(1), 3.50(1) |
| (bant)₂(14tfib)₅ | −5.54 | −19.48 (−19.13, −20.29, −18.30, −17.45, −22.25) | 3.50(2), 3.64(2), 3.50(1), 3.46(1), 3.51(3), 3.39(2), 3.48(2), 3.40(2), 3.55(1) |
| (pyr)(14tfib)₂ | −8.46 | −20.45 (−20.45, −20.45) | 3.509(9), 3.430(8), 3.482(9), 3.51(1) |
| (cor)(14tfib)₂ | −5.93 | −20.45 (−20.45, −20.45) | 3.45(1), 3.46(2), 3.46(1), 3.51(2) |
| (dicor)(14tfib)₃ | 6.48 | −20.12 (−19.54, −20.29, −20.55) | 3.440(8), 3.424(7), 3.368(8), 3.38(1), 3.560(9) |
| (anthra)(ofiab)₂ | −2.91 | −19.66 (−19.30, −20.31) | 3.62(2), 3.43(1), 3.55(2), 3.63(2) |
| (pyr)(ofiab)₂ | −0.17 | −20.33 (−20.00, −20.66) | 3.476(7), 3.394(7), 3.586(7), 3.448(6) |
| (C₆₀)(14tfib)₂ | −13.95 | −15.15 | 3.54(1) |

^a The full set of XB bonds and angles is provided in ESI Table S2. ^b Calculation based on the orthorhombic polymorph of **benz** (CSD code BENZEN).



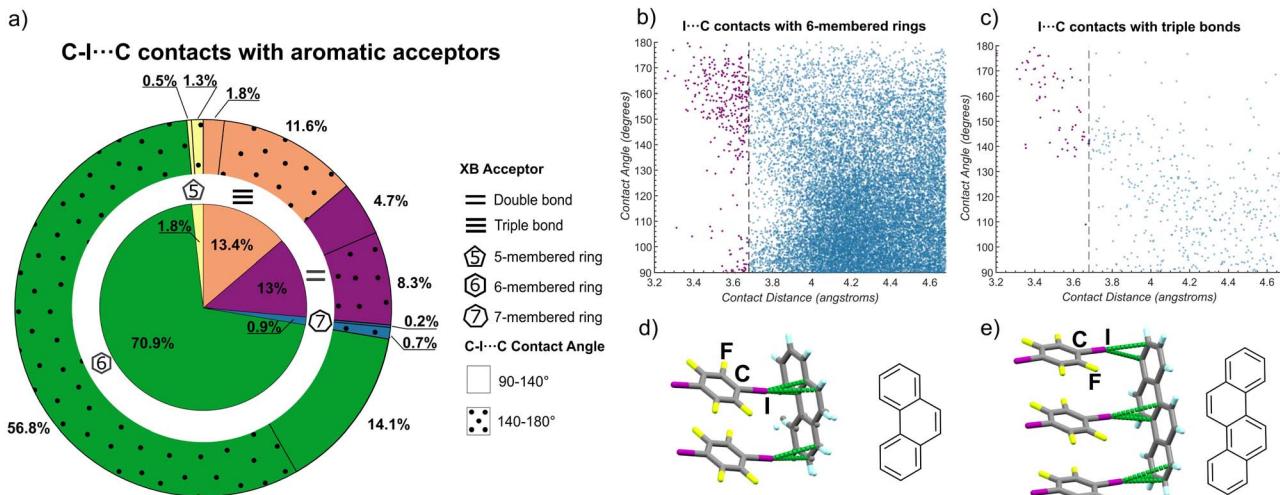


Fig. 4 Outcomes of CSD searches for short C-I...C contacts to alkene, alkyne, 5-, 6- and 7-membered ring acceptors based on aromatic carbon and selected representative structures. (a) Distribution of the I...C contact lengths shorter than 4.68 Å across different acceptor types. For each XB acceptor type, the fraction of contacts with a linear geometry (C-I...C angle in the range from 140° to 180°) corresponding to halogen bonding, is indicated on the outer circle by a dotted pattern. Plots of the distribution of lengths and angles for C-I...C contacts to: (b) 6-membered ring and (c) alkyne (C≡C) acceptors. To guide the eye, contacts shorter than the van der Waals limit of 3.68 Å are shown in purple. Plots of the distribution of C-I...C contact lengths and angles for 5- and 7-membered rings, as well as alkene (C=C) acceptors, are given in the ESI.† Fragment of the crystal structures⁶⁷ of: (d) (phenanthrene)(14tfib)₂ and (e) (chrysene)(14tfib)₃, illustrating the appearance of the C-I...π_C ladder motif.

Periodic DFT was previously shown⁶⁶ to be effective for understanding the thermodynamic stability of halogen-bonded co-crystals. The calculated co-crystal formation energies (E_{calc}) with respect to individual crystalline components (Table 1) were found to be negative in all but one case, indicating that co-crystallisation was enthalpically favorable in general. The exception is **(dicor)(14tfib)₃**, with E_{calc} of +6.48 kJ mol⁻¹ suggesting that its formation may be entropy driven. Vibrational contributions to the Free energies related to co-crystallisation were, however, not taken into account due to the high computational cost of periodic DFT phonon calculations.

The energies associated with individual halogen bonds (E_{XB}) were obtained by subtracting the total energies of individual component molecules from the total energy of each distinct XB donor-acceptor dimer unit found in the co-crystal structure. The E_{XB} values were highly consistent across all explored XB acceptors, with all dimer interaction energies being exothermic and falling within the range of -13 kJ mol⁻¹ to -21 kJ mol⁻¹.

Analysis of the Cambridge structural database (CSD)

Our systematic experimental study reveals the robust supramolecular synthon based on C-I...π_C halogen bonds for the directional assembly of carbon-only aromatic systems of various shapes and sizes. It is highly surprising that this expansive potential for directional arene assembly by C-I...π_C bonds has not previously been noted, leading us to conduct an in-depth search of the Cambridge Structural Database (CSD) to probe for the appearance and geometry of such interactions involving the iodine atom of a C-I fragment as a donor and a carbon atom of a 5-, 6- or a 7-membered ring, a C=C, or a C≡C moiety, as the acceptor. As XB are expected to be shorter than the sum of the

van der Waals radii of interacting atoms, the search was limited to I...C distances up to 4.68 Å, which is *ca.* 1 Å longer than the sum of van der Waals radii for carbon (1.7 Å) and iodine (1.98 Å).⁴⁹

The CSD searches revealed that short contacts between iodine and 6-membered rings are the most prevalent, representing *ca.* 71% of the total short contacts examined, followed by contacts to C≡C bonds (13%), C=C bonds (13%), while contacts involving 5- and 7-membered rings represented *ca.* 2% and 1% of all identified short contacts (Fig. 4a). The CSD searches indicated two geometry-distinct regimes of C-I...π_C interactions, dependent on the I...C_π distance. Specifically, there was no particular geometrical preference for longer I...C separations. However, shorter I...C distances were generally associated with a preference for linear C-I...C contacts, with angles in the range of *ca.* 140–180°. Such behavior was particularly pronounced for C≡C bonds and 6-membered rings as acceptors, where the majority of I...C distances below *ca.* 3.7 Å were associated with a C-I...C angle of 140° or higher (Fig. 4b and c). Such preference for linear geometries at lower I...C distances is consistent with XB^{14d} reinforcing the herein proposed role of halogen bonds as an overlooked, yet consistent, strategy for directional assembly of carbon-based aromatic system, and suggest the possible existence of other C-I...π_C driven supramolecular synthons. In addition to structures identified in the CSD, during preparation of this manuscript two crystal structures were reported⁶⁷ that strongly reinforce the view of XB as reliable interactions to form supramolecular architectures based on PAHs (Fig. 4d and e). These structures are based on phenanthrene and on chrysene, an isomer of **tet** and **bant**, and exhibit the C-I...π_C ladder motif with PAH as the XB acceptor and **14tfib** as the donor.



Luminescence properties

Organic materials with emissive triplet states and long emission lifetimes (phosphorescence) are being sought for applications in OLEDs,⁶⁸ biological imaging,⁶⁹ anticounterfeiting⁷⁰ and more.⁷¹ However, purely organic systems rarely exhibit phosphorescence at room temperature, due to the spin-forbidden nature of the singlet-to-triplet transition.^{72–74} Recent work shows the critical roles of intermolecular interactions and crystal packing on organic room temperature phosphorescence efficiency.^{74–76} In this context, cocrystallisation of PAHs offers a promising platform for achieving organic phosphorescent materials, as the solid-state arrangement of aromatic units directly influences diverse optoelectronic properties, such as excitation/emission wavelengths, and lifetime kinetics.^{77,78} Halogen bonded arene cocrystals have been reported to exhibit phosphorescence decays on the scale of several milliseconds,^{27–32,42,67,77} with an isolated case of decays >200 ms.^{77b} Recently, Abe *et al.* reported luminescence properties of a (phenanthrene)(14tfib)₂ cocrystal, and found that partial replacement (up to 25%) of phenanthrene with **pyr** leads to an increase of the phosphorescence quantum yield from *ca.* 6 to >20%.⁶⁷ Further replacement of phenanthrene with **pyr** (up to 50%), however, led to a drop of phosphorescence efficiency. Notably, a cocrystal of composition (**pyr**)(14tfib)₂, in which all phenanthrene would be replaced by **pyr**, was not reported – but (**pyr**)(14tfib) showed a low luminescence quantum yield <1% and short lifetime of 50 μs⁶⁷ (500 μs²⁹).

Having herein synthesised the previously missing cocrystal of composition (**pyr**)(14tfib)₂, we explored its photoluminescence properties, along with those of the heavier congeners (**pery**)(14tfib)₂ and (**cor**)(14tfib)₂ (Fig. 5). The cocrystals of **cor**, **pyr**, or **pery** with 14tfib all exhibited strong luminescence shifts when compared to the pristine solid PAH. The previously not accessible (**pyr**)(14tfib)₂ exhibited a remarkable bathochromic shift of almost 200 nm, with the maximum emission wavelength (λ_{max}) at 678 nm (red), compared to 498 nm (blue) for solid **pyr** (Fig. 5a and b). The red-shifted λ_{max} and the emission life-time (10.4 μs) are similar to the previously reported behavior of (**pyr**)(14tfib)^{29,67} and suggest the phosphorescent nature of the emission. A similar behavior was displayed by (**cor**)(14tfib)₂ cocrystals: a significant ~150 nm red shift of λ_{max} (from 513 nm in pristine solid **cor** to 660 nm in the cocrystal, Fig. 5c and d) and very long emission life-time (4.2 ms). Notably, the emission of (**cor**)(14tfib)₂ extends in the NIR region, beyond 750 nm, which is typically difficult to achieve because of the energy-gap law.⁷⁸

In a sharp contrast to the red-shifted emission (**pyr**)(14tfib)₂ and (**cor**)(14tfib)₂ cocrystals, the (**pery**)(14tfib)₂ cocrystals exhibited a strong ~100 nm hypsochromic shift of emission, from $\lambda_{\text{max}} = 607$ nm (orange) in pristine **pery** solid (α -**pery** polymorph) to 521 nm (green) of the cocrystal (Fig. 5e and f). Such blue-shifting behavior is unusual, as cocrystals exhibiting C–I \cdots π_{C} interactions generally exhibit red-shifted emission compared to the pure emissive component solid.⁷⁹ This behavior of (**pery**)(14tfib)₂ is reminiscent of the J-aggregate emission of the β -**pery** crystals⁸⁰ in which the **pery** molecules are arranged in off-diagonal

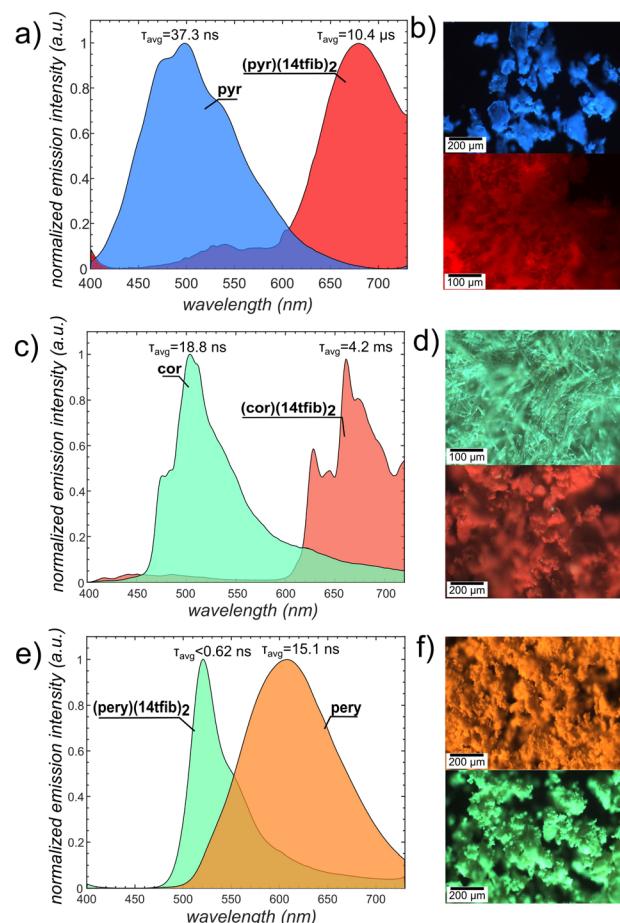


Fig. 5 Change in luminescence spectra upon the formation of halogen-bonded cocrystals with aromatic XB acceptors. (a) Solid-state luminescence spectra for **pyr** (blue) and (**pyr**)(14tfib)₂ (red). (b) A sample of (top) **pyr** and (bottom) (**pyr**)(14tfib)₂ excited with 365 nm LED light. (c) Solid-state luminescence spectra for **cor** (green) and (**cor**)(14tfib)₂ (red). (d) A sample of (top) **cor** and (bottom) (**cor**)(14tfib)₂ excited with 365 nm LED light. (e) Solid-state luminescence spectra for α -**pery** and (**pery**)(14tfib)₂. (f) A sample of (top) α -**pery** and (bottom) (**pery**)(14tfib)₂ excited with 365 nm LED light.

interaggregate interactions, similar to (**pery**)(14tfib)₂ (see ESI, Fig. S36†). Such J-aggregated states are known to enhance the fluorescence emission radiative rates, outcompeting intersystem crossing rates from populating triplet excitons.⁸¹ Indeed, the emission lifetime observed in (**pery**)(14tfib)₂ was <0.62 ns (see ESI, Fig. S35†). The observed blue shift of (**pery**)(14tfib)₂ cocrystals (as well as β -**pery** crystals) compared to α -**pery** is explained by disruption of the excimer emission resulting from the π -dimer packing motif of the latter structure.

Consequently, the incorporation of the C–I \cdots π_{C} interactions in co-crystals of the PAHs can enable their room-temperature phosphorescence (RTP) through the heavy atom effect, and is commonly manifested in the red-shifted emission of cocrystals compared to pure PAH. However, the RTP properties also depend on a balance of inter-system crossing *vs.* fluorescence transition which varies with the molecular and crystal structure. Thus, the fast fluorescence transition in (**pery**)(14tfib)₂ brought

about by the specific packing of PAH molecules (J-aggregation) prevents the observation of RTP.

Conclusions

In summary, we have presented an extensive and systematic experimental, theoretical, and database study that establishes halogen bonding to aromatic carbon systems as a powerful, but historically overlooked approach for the design of supramolecular architectures based on carbon as the molecular recognition site. This work coalesces and advances scattered reports of halogen bonds to π -systems, and demonstrates not only that C–I \cdots π_C halogen bonding can be reliably used as a directional interaction for the synthesis of cocrystals of differently-sized and -shaped carbon-only aromatic systems, but also establishes the supramolecular ladder structure based on C–I \cdots π_C halogen bonds as a unique supramolecular synthon generally applicable to non-substituted carbon systems.

In particular, C–I \cdots π_C halogen bonding has been effective in cocrystallisation of all 9 explored aromatic carbon molecules, from benzene (one aromatic ring) to diconorylene (15 fused aromatic rings), resulting in altogether 11 cocrystals. Focusing on PAHs only (*i.e.*, excluding benzene and C₆₀), C–I \cdots π_C halogen bonding led to the formation of the expected supramolecular ladder motif in 8 out of 9 cases, with PAH units acting as the rungs and halogen bond donor molecules as the rails. The reliability of a self-assembly motif can be evaluated through Aakeröy's concept of supramolecular yield.^{44–46} From that perspective, the synthesis of C–I \cdots π_C ladder motifs based on non-substituted arene hydrocarbons is highly reliable, with an 89% supramolecular yield. If one includes in this overview the recently reported cocrystals of naphthalene,³⁰ azulene (two cocrystals),⁴³ phenanthrene and chrysene,⁶⁷ which were all found to exhibit the C–I \cdots π_C ladder motif, the supramolecular yield rises to 13 out of 14 cases, *i.e.* 93%.

The high reliability of this supramolecular cocrystal-forming reaction is surprising when considering that traditionally the only reliable way to achieve cocrystallisation of arenes was through π -stacking. The C–I \cdots π_C halogen-bonded ladder now suggests a route for crystal engineers to rationally use non-substituted PAHs as building blocks in designing new carbon-based supramolecular architectures, different from π -stacked arrays.[‡] The C–I \cdots π_C halogen bonds underlying the ladder motif often coincide with the sites of greatest negative ESP on the arene, and a single side of a six-membered carbon ring is never observed to participate in more than one halogen bond. These observations establish a different element of design for the assembly of supramolecular architectures from non-substituted arenes, based on the directional interaction of halogen bonding, rather than the traditionally used and non-directional π -stacking. The ability to generate new structures based on simple PAH components is of significance in the context of materials with new properties. That is illustrated by very large red or blue shifts (in the range of *ca.* 100 nm to 200 nm), as well as room-temperature phosphorescence achieved for PAHs engaged in the supramolecular C–I \cdots π_C ladder structures. In particular, the phosphorescence lifetime of >4 ms

observed for the coronene-based cocrystal provides further incentive to explore the design of fully organic phosphorescent systems based on halogen bonding to aromatic system.

Data availability

Data supporting this manuscript has been provided as ESI.[†] Selected data relevant to PXRD analysis, fluorescence emission spectroscopy, and theoretical modelling, is also available from the data repository Zenodo (DOI: <https://doi.org/10.5281/zenodo.10064165>). Crystallographic data in CIF format has been deposited with the Cambridge Structural Database (CCDC 2240199–2240207, 2249511 and 2281269).

Author contributions

The manuscript was written through contributions of all authors. All authors have given approval to the final version of the manuscript.

Conflicts of interest

There are no conflicts to declare.

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Notes and references

[†] d'Agostino and co-workers⁴² have reported that the aromatic hydrocarbons stilbene and tolane form halogen-bonded cocrystals with **14tfb** which, although they do not exhibit the C–I \cdots π_C ladder synthon, are based on a closely-related sheet architecture.

- 1 E. Peeters and J. Cami, in *Encyclopedia of Astrobiology*, ed. M. Gargaud, R. Amils, J. C. Quintanilla, H. J. Cleaves, W. M. Irvine, D. L. Pinti and M. Viso, Springer Berlin Heidelberg, Berlin, Heidelberg, 2011, ch. 1250, pp. 1307–1321.
- 2 E. Peeters, C. Mackie, A. Candian and A. G. G. M. Tielens, *Acc. Chem. Res.*, 2021, **54**, 1921–1933.
- 3 C. S. Hansen, E. Peeters, J. Cami and T. W. Schmidt, *Commun. Chem.*, 2022, **5**, 94.
- 4 C. Aumaitre and J. F. Morin, *Chem. Rec.*, 2019, **19**, 1142–1154.
- 5 J. Zaumseil and H. Sirringhaus, *Chem. Rev.*, 2007, **107**, 1296–1323.



6 J. H. Lee, C. H. Chen, P. H. Lee, H. Y. Lin, M. K. Leung, T. L. Chiu and C. F. Lin, *J. Mater. Chem. C*, 2019, **7**, 5874–5888.

7 N. Liang, Y. K. Zhao, Y. Z. Wu, C. R. Zhang and M. Shao, *Appl. Phys. Lett.*, 2021, **119**, 053301.

8 X. Yang, L. Lan, X. Pan, X. Liu, Y. Song, X. Yang, Q. Dong, L. Li, P. Naumov and H. Zhang, *Nat. Commun.*, 2022, **13**, 7874.

9 (a) P. P. Yu, Y. G. Zhen, H. L. Dong and W. P. Hu, *Chem*, 2019, **5**, 2814–2853; (b) P. Vishweshwar, J. A. McMahon, J. A. Bis and M. J. Zaworotko, *J. Pharm. Sci.*, 2006, **95**, 499–516; (c) G. R. Desiraju, *J. Am. Chem. Soc.*, 2013, **135**, 9952–9967; (d) M. K. Corpinot and D.-K. Bučar, *Cryst. Growth Des.*, 2019, **19**, 1426–1453; (e) D. Braga, F. Grepioni and G. R. Desiraju, *Chem. Rev.*, 1998, **98**, 1375–1406; (f) L. Sun, W. Zhu, X. Zhang, L. Li, H. Dong and W. Hu, *J. Am. Chem. Soc.*, 2021, **143**, 19243–19256.

10 (a) J. E. Anthony, *Chem. Rev.*, 2006, **106**, 5028–5048; (b) B. Strehmel, A. M. Sarker, J. H. Malpert, V. Strehmel, H. Seifert and D. C. Neckers, *J. Am. Chem. Soc.*, 1999, **121**, 1226–1236; (c) Z. Chen, P. Müller and T. M. Swager, *Org. Lett.*, 2006, **8**, 273–276; (d) S. E. Wheeler, *J. Am. Chem. Soc.*, 2011, **133**, 10262–10274.

11 P. Gupta, D. P. Karothu, E. Ahmed, P. Naumov and N. K. Nath, *Angew. Chem., Int. Ed.*, 2018, **57**, 8498–8502.

12 S. P. Yelgaonkar, G. Campillo-Alvarado and L. R. MacGillivray, *J. Am. Chem. Soc.*, 2020, **142**, 20772–20777.

13 G. R. Krishna, R. Devarapalli, G. Lal and C. M. Reddy, *J. Am. Chem. Soc.*, 2016, **138**, 13561–13567.

14 (a) C. B. Aakeröy, M. Baldighi, J. Desper, P. Metrangolo and G. Resnati, *Chem.-Eur. J.*, 2013, **19**, 16240–16247; (b) C. Gilday, S. W. Robinson, T. A. Barendt, M. J. Langton, B. R. Mullaney and P. D. Beer, *Chem. Rev.*, 2015, **115**, 7118–7195; (c) J. Teyssandier, K. S. Mali and S. De Feyter, *ChemistryOpen*, 2020, **9**, 225–241; (d) G. Cavallo, P. Metrangolo, R. Milani, T. Pilati, A. Priimagi, G. Resnati and G. Terraneo, *Chem. Rev.*, 2016, **116**, 2478–2601; (e) G. R. Desiraju, P. S. Ho, L. Kloof, A. C. Legon, R. Marquardt, P. Metrangolo, P. Politzer, G. Resnati and K. Rissanen, *Pure Appl. Chem.*, 2013, **85**, 1711–1713; (f) N. Juneja, N. M. Shapiro, D. K. Unruh, E. Bosch, R. H. Groeneman and K. M. Hutchins, *Angew. Chem., Int. Ed.*, 2022, **61**, e202202708; (g) M. A. Sinnwell and L. R. MacGillivray, *Angew. Chem., Int. Ed.*, 2016, **55**, 3477–3480; (h) R. Gutzler, C. Fu, A. Dadvand, Y. Hua, J. M. MacLeod, F. Rosei and D. F. Perepichka, *Nanoscale*, 2012, **4**, 5965–5971.

15 (a) G. Ferguson, J. F. Gallagher, C. Glidewell and C. M. Zakaria, *Acta Crystallogr., Sect. C: Struct. Chem.*, 1994, **50**, 70–73; (b) M. Nishio, *Phys. Chem. Chem. Phys.*, 2011, **13**, 13873–13900; (c) J. L. Atwood, F. Hamada, K. D. Robinson, G. W. Orr and R. L. Vincent, *Nature*, 1991, **349**, 683–684; (d) K. Fucke, K. M. Anderson, M. H. Filby, M. Henry, J. Wright, S. A. Mason, M. J. Gutmann, L. J. Barbour, C. Oliver, A. W. Coleman, J. L. Atwood, J. A. Howard and J. W. Steed, *Chem.-Eur. J.*, 2011, **17**, 10259–10271; (e) F. H. Allen, V. J. Hoy, J. A. K. Howard, V. R. Thalladi, G. R. Desiraju, C. C. Wilson and G. J. McIntyre, *J. Am. Chem. Soc.*, 1997, **119**, 3477–3480.

16 U. Geiser, S. K. Kumar, B. M. Savall, S. S. Harried, K. D. Carlson, P. R. Mobley, H. H. Wang, J. M. Williams and R. E. Botto, *Chem. Mater.*, 1992, **4**, 1077–1082.

17 N. R. Goud, O. Bolton, E. C. Burgess and A. J. Matzger, *Cryst. Growth Des.*, 2016, **16**, 1765–1771.

18 D. Cao, M. Hong, A. K. Blackburn, Z. Liu, J. M. Holcroft and J. F. Stoddart, *Chem. Sci.*, 2014, **5**, 4242–4248.

19 (a) H. J. Cho, S. W. Kim, S. Kim, S. Lee, J. Lee, Y. Cho, Y. Lee, T.-W. Lee, H.-J. Shin and C. Song, *J. Mater. Chem. C*, 2020, **8**, 17289–17296; (b) G. W. Coates, A. R. Dunn, L. M. Henling, J. W. Ziller, E. B. Lobkovsky and R. H. Grubbs, *J. Am. Chem. Soc.*, 1998, **120**, 3641–3649; (c) A. Friedrich, I. E. Collings, K. F. Dziubek, S. Fanetti, K. Radacki, J. Ruiz-Fuertes, J. Pellicer-Porres, M. Hanfland, D. Sieh, R. Bini, S. J. Clark and T. B. Marder, *J. Am. Chem. Soc.*, 2020, **142**, 18907–18923; (d) J. C. Collings, K. P. Roscoe, E. G. Robins, A. S. Batsanov, L. M. Stimson, J. A. K. Howard, S. J. Clark and T. B. Marder, *New J. Chem.*, 2002, **26**, 1740–1746; (e) L. Wang, J. Deng, M. Jiang, C. Zhen, F. Li, S. Li, S. Bai, X. Zhang and W. Zhu, *J. Mater. Chem. A*, 2023, **11**, 11235–11244; (f) B. L. Hu and Q. Zhang, *Chem. Rec.*, 2021, **21**, 116–132; (g) T. Faisal, K. M. Solntsev, T. Kahs, N. Saleh, P. Commins, J. Whelan, S. Mohamed and P. Naumov, *Energy Fuels*, 2021, **35**, 8742–8755; (h) Z. M. Saeed, B. Dhokale, A. F. Shunnar, W. M. Awad, H. H. Hernandez, P. Naumov and S. Mohamed, *Cryst. Growth Des.*, 2021, **21**, 4151–4161; (i) G. Raj, Y. Kikkawa, L. Catalano, R. Pasricha, Y. Norikane and P. Naumov, *Chemnanomat*, 2020, **6**, 68–72; (j) T. Schillmöller, R. Herbst-Irmer and D. Stalke, *Adv. Opt. Mater.*, 2021, **9**, 2001814; (k) N. G. Petrov, P. Chartier, T. Maris and J. D. Wuest, *J. Am. Chem. Soc.*, 2022, **144**, 556–572; (l) K. Biradha and R. Santra, *Chem. Soc. Rev.*, 2013, **42**, 950–967; (m) M. C. Gerthoffer, B. Xu, S. Wu, J. Cox, S. Huss, S. M. Oburn, S. A. Lopez, V. H. Crespi, J. V. Badding and E. Elacqua, *Polym. Chem.*, 2022, **13**, 1359–1368; (n) A. J. Oyer, J.-M. Y. Carrillo, C. C. Hire, H. C. Schniepp, A. D. Asandei, A. V. Dobrynin and D. H. Adamson, *J. Am. Chem. Soc.*, 2012, **134**, 5018–5021; (o) Y. Xu and M. von Delius, *Angew. Chem., Int. Ed.*, 2020, **59**, 559–573; (p) M. Bartkowiak and S. Giordani, *Nanoscale*, 2020, **12**, 9352–9358.

20 L. R. MacGillivray, *CrystEngComm*, 2004, **6**, 77–78.

21 C. A. Gunawardana and C. B. Aakeröy, *Chem. Commun.*, 2018, **54**, 14047–14060.

22 A. K. Nangia and G. R. Desiraju, *Angew. Chem., Int. Ed.*, 2019, **58**, 4100–4107.

23 (a) D. Cinčić, T. Friščić and W. Jones, *J. Am. Chem. Soc.*, 2008, **130**, 7524–7525; (b) D. Cinčić, T. Friščić and W. Jones, *CrystEngComm*, 2011, **13**, 3224–3231; (c) Y. Xu, J. Huang, B. Gabidullin and D. L. Bryce, *Chem. Commun.*, 2018, **54**, 11041–11043; (d) A. A. Eliseeva, D. M. Ivanov, A. V. Rozhkov, I. V. Ananyev, A. Frontera and V. Y. Kukushkin, *JACS Au*, 2021, **1**, 354–361; (e) A. Bauza, D. Quinonero, A. Frontera and P. M. Deya, *Phys. Chem. Chem. Phys.*, 2011, **13**, 20371–20379; (f) E. A. Katlenok,



A. V. Rozhkov, O. V. Levin, M. Haukka, M. L. Kuznetsov and V. Y. Kukushkin, *Cryst. Growth Des.*, 2021, **21**, 1159–1177; (g) I. Benito, R. M. Gomila and A. Frontera, *CrystEngComm*, 2022, **24**, 4440–4446; (h) K. Lisac, F. Topić, M. Arhangelskis, S. Cepic, P. A. Julien, C. W. Nickels, A. J. Morris, T. Friščić and D. Cinčić, *Nat. Commun.*, 2019, **10**, 61; (i) P. J. Costa, *Phys. Sci. Rev.*, 2017, **2**; (j) T. H. Borchers, F. Topić, J. C. Christopherson, O. S. Bushuyev, J. Vainauskas, H. M. Titi, T. Friščić and C. J. Barrett, *Nat. Chem.*, 2022, **14**, 574–581.

24 X. H. Ding, Y. Z. Chang, C. J. Ou, J. Y. Lin, L. H. Xie and W. Huang, *Natl. Sci. Rev.*, 2020, **7**, 1906–1932.

25 S. A. Sharber, W. J. Mullin and S. W. Thomas, *Chem. Mater.*, 2021, **33**, 6640–6661.

26 R. Liantonio, S. Luzzati, P. Metrangolo, T. Pilati and G. Resnati, *Tetrahedron*, 2002, **58**, 4023–4029.

27 W. Z. Wang, Y. Zhang and W. J. Jin, *Coord. Chem. Rev.*, 2020, **404**, 213107.

28 L. L. Li, Z. F. Liu, W. X. Wu and W. J. Jin, *Acta Crystallogr., Sect. B: Struct. Sci., Cryst. Eng. Mater.*, 2018, **74**, 610–617.

29 Q. J. Shen, H. Q. Wei, W. S. Zou, H. L. Sun and W. J. Jin, *CrystEngComm*, 2012, **14**, 1010–1015.

30 Q. J. Shen, X. Pang, X. R. Zhao, H. Y. Gao, H.-L. Sun and W. J. Jin, *CrystEngComm*, 2012, **14**, 5027–5034.

31 H. Y. Gao, X. R. Zhao, H. Wang, X. Pang and W. J. Jin, *Cryst. Growth Des.*, 2012, **12**, 4377–4387.

32 H. Y. Gao, Q. J. Shen, X. R. Zhao, X. Q. Yan, X. Pang and W. J. Jin, *J. Mater. Chem.*, 2012, **22**, 5336–5343.

33 L. Li, H. Wang, W. Wang and W. J. Jin, *CrystEngComm*, 2017, **19**, 5058–5067.

34 L. Li, W. X. Wu, Z. F. Liu and W. J. Jin, *New J. Chem.*, 2018, **42**, 10633–10641.

35 S. J. Ang, A. M. Mak, M. B. Sullivan and M. W. Wong, *Phys. Chem. Chem. Phys.*, 2018, **20**, 8685–8694.

36 A. Forni, S. Pieraccini, S. Rendine, F. Gabas and M. Sironi, *ChemPhysChem*, 2012, **13**, 4224–4234.

37 D. Y. Kim, J. M. L. Madridejos, M. Ha, J. H. Kim, D. C. Yang, C. Baig and K. S. Kim, *Chem. Commun.*, 2017, **53**, 6140–6143.

38 T. J. Mooibroek and P. Gamez, *CrystEngComm*, 2013, **15**, 1802–1805.

39 Y. Zhang, J. G. Wang, X. Sun, Q. Liu, W. Wang and Y. B. Wang, *ChemPlusChem*, 2018, **83**, 470–477.

40 A. S. Mikherdov, A. S. Novikov, V. P. Boyarskiy and V. Y. Kukushkin, *Nat. Commun.*, 2020, **11**, 2921.

41 A. S. Mikherdov, R. A. Popov, A. S. Smirnov, A. A. Eliseeva, A. S. Novikov, V. P. Boyarskiy, R. M. Gomila, A. Frontera, V. Y. Kukushkin and N. A. Bokach, *Cryst. Growth Des.*, 2022, **22**, 6079–6087; A. S. Smirnov, E. A. Katlenok, A. S. Mikherdov, M. A. Kryukova, N. A. Bokach and V. Y. Kukushkin, *Int. J. Mol. Sci.*, 2023, **24**, 13324.

42 (a) S. d'Agostino, F. Grepioni, D. Braga and B. Ventura, *Cryst. Growth Des.*, 2015, **15**, 2039–2045; (b) G. Lapadula, N. Judaš, T. Friščić and W. Jones, *Chem.–Eur. J.*, 2010, **16**, 7400–7403; (c) S. Y. Oh, C. W. Nickels, F. Garcia, W. Jones and T. Friščić, *CrystEngComm*, 2012, **14**, 6110–6114.

43 J. Vainauskas, F. Topić, O. S. Bushuyev, C. J. Barrett and T. Friščić, *Chem. Commun.*, 2020, **56**, 15145–15148.

44 C. B. Aakeröy, A. M. Beatty and B. A. Helfrich, *J. Am. Chem. Soc.*, 2002, **124**, 14425–14432.

45 C. B. Aakeröy, A. M. Beatty and B. A. Helfrich, *Angew. Chem. Int. Ed. Engl.*, 2001, **40**, 3240–3242.

46 C. B. Aakeröy and D. J. Salmon, *CrystEngComm*, 2005, **7**, 439–448.

47 G. R. Desiraju, *Angew. Chem. Int. Ed. Engl.*, 1995, **34**, 2311–2327.

48 C. R. Groom, I. J. Bruno, M. P. Lightfoot and S. C. Ward, *Acta Crystallogr., Sect. B: Struct. Sci., Cryst. Eng. Mater.*, 2016, **72**, 171–179.

49 M. Mantina, A. C. Chamberlin, R. Valero, C. J. Cramer and D. G. Truhlar, *J. Phys. Chem. A*, 2009, **113**, 5806–5812.

50 O. Hassel, K. O. Strømme, H. Haraldsen, A. Grönvall, B. Zaar and E. Diczfalusy, *Acta Chem. Scand.*, 1958, **12**, 1146.

51 E. Bosch, *IUCrData*, 2019, **4**, x190993.

52 I. Huskić, J. C. Christopherson, K. Užarević and T. Friščić, *Chem. Commun.*, 2016, **52**, 5120–5123.

53 T. Friščić, S. L. Childs, S. A. A. Rizvi and W. Jones, *CrystEngComm*, 2009, **11**, 418–426.

54 T. Friščić, A. V. Trask, W. Jones and W. D. Motherwell, *Angew. Chem. Int. Ed. Engl.*, 2006, **45**, 7546–7550.

55 A. Azzali, S. d'Agostino, M. Capacci, F. Spinelli, B. Ventura and F. Grepioni, *CrystEngComm*, 2022, **24**, 5748–5756.

56 H. Jain, D. Sutradhar, S. Roy and G. R. Desiraju, *Angew. Chem. Int. Ed. Engl.*, 2021, **60**, 12841–12846.

57 M. Zander and W. Franke, *Chem. Ber./Recl.*, 1958, **91**, 2794–2797.

58 Z. Mu, L. Shu, H. Fuchs, M. Mayor and L. Chi, *J. Am. Chem. Soc.*, 2008, **130**, 10840–10841.

59 Y. Quo, N. Karasawa and W. A. Goddard, *Nature*, 1991, **351**, 464–467.

60 D. André, A. Dworkin, H. Szwarc, R. Céolin, V. Agafonov, C. Fabre, A. Rassat, L. Straver, P. Bernier and A. Zahab, *Mol. Phys.*, 1992, **76**, 1311–1317.

61 A. Giri and P. E. Hopkins, *J. Phys. Chem. Lett.*, 2017, **8**, 2153–2157.

62 T. Matsuno, Y. Nakai, S. Sato, Y. Maniwa and H. Isobe, *Nat. Commun.*, 2018, **9**, 1907.

63 S. J. Clark, M. D. Segall, C. J. Pickard, P. J. Hasnip, M. J. Probert, K. Refson and M. C. Payne, *Z. Kristallogr. Cryst. Mater.*, 2005, **220**, 567–570.

64 J. P. Perdew, K. Burke and M. Ernzerhof, *Phys. Rev. Lett.*, 1996, **77**, 3865–3868.

65 S. Ehrlich, J. Moellmann, W. Reckien, T. Bredow and S. Grimme, *ChemPhysChem*, 2011, **12**, 3414–3420.

66 (a) K. Lisac, V. Nemec, F. Topić, M. Arhangelskis, P. Hindle, R. Tran, I. Huskić, A. J. Morris, T. Friščić and D. Cinčić, *Cryst. Growth Des.*, 2018, **18**, 2387–2396; (b) F. Topić, K. Lisac, M. Arhangelskis, K. Rissanen, D. Cinčić and T. Friščić, *Chem. Commun.*, 2019, **55**, 14066–14069; (c) M. Arhangelskis, F. Topić, P. Hindle, R. Tran, A. J. Morris, D. Cinčić and T. Friščić, *Chem. Commun.*, 2020, **56**, 8293–8296; (d) L. Kumar, K. Leko, V. Nemec, D. Trzybinski, N. Bregovic, D. Cinčić and M. Arhangelskis, *Chem. Sci.*, 2023, **14**, 3140–3146.



67 A. Abe, K. Goushi, M. Mamada and C. Adachi, *Adv. Mater.*, 2023, e2211160.

68 G. Zhan, Z. Liu, Z. Bian and C. Huang, *Front. Chem.*, 2019, **7**, 305.

69 J. Yang, X. Zhen, B. Wang, X. Gao, Z. Ren, J. Wang, Y. Xie, J. Li, Q. Peng, K. Pu and Z. Li, *Nat. Commun.*, 2018, **9**, 840.

70 Y. Su, S. Z. F. Phua, Y. Li, X. Zhou, D. Jana, G. Liu, W. Q. Lim, W. K. Ong, C. Yang and Y. Zhao, *Sci. Adv.*, 2018, **4**, eaas9732.

71 W. J. Zhao, Z. K. He and B. Z. Tang, *Nat. Rev. Mater.*, 2020, **5**, 869–885.

72 Kenry, C. Chen and B. Liu, *Nat. Commun.*, 2019, **10**, 2111.

73 E. Hamzehpoor, C. Ruchlin, Y. Tao, C. H. Liu, H. M. Titi and D. F. Perepichka, *Nat. Chem.*, 2023, **15**, 83–90.

74 Z. An, C. Zheng, Y. Tao, R. Chen, H. Shi, T. Chen, Z. Wang, H. Li, R. Deng, X. Liu and W. Huang, *Nat. Mater.*, 2015, **14**, 685–690.

75 J. Yang, X. Zhen, B. Wang, X. Gao, Z. Ren, J. Wang, Y. Xie, J. Li, Q. Peng, K. Pu and Z. Li, *Nat. Commun.*, 2018, **9**, 840.

76 E. Hamzehpoor and D. F. Perepichka, *Angew. Chem., Int. Ed.*, 2020, **59**, 9977–9981.

77 (a) H. Wang, R. X. Hu, X. Pang, H. Y. Gao and W. J. Jin, *CrystEngComm*, 2014, **16**, 7942–7948; (b) S. d'Agostino, F. Spinelli, P. Taddei, B. Ventura and F. Grepioni, *Cryst. Growth Des.*, 2018, **19**, 336–346; (c) M. Singh, K. Liu, S. L. Qu, H. L. Ma, H. F. Shi, Z. F. An and W. Huang, *Adv. Opt. Mater.*, 2021, **9**, 2002197.

78 Y. C. Wei, S. F. Wang, Y. Hu, L. S. Liao, D. G. Chen, K. H. Chang, C. W. Wang, S. H. Liu, W. H. Chan, J. L. Liao, W. Y. Hung, T. H. Wang, P. T. Chen, H. F. Hsu, Y. Chi and P. T. Chou, *Nat. Photonics*, 2020, **14**, 570–577.

79 Y. Huang, Z. Wang, Z. Chen and Q. Zhang, *Angew Chem. Int. Ed. Engl.*, 2019, **58**, 9696–9711.

80 K. Sato and R. Kato, *Chem. Phys. Lett.*, 2019, **730**, 312–315.

81 N. J. Hestand and F. C. Spano, *Chem. Rev.*, 2018, **118**, 7069–7163.

