

Cite this: *Chem. Sci.*, 2023, 14, 10953

All publication charges for this article have been paid for by the Royal Society of Chemistry

A Pd-based plasmonic photocatalyst for nitrogen fixation through an antenna–reactor mechanism†

Yuanyuan Yang,^{‡a} Henglei Jia,^{‡*a} Sihua Su,^c Yidi Zhang,^a Mengxuan Zhao,^a Jingzhao Li,^a Qifeng Ruan^{‡*c} and Chun-yang Zhang^{‡*ab}

Plasmonic metal nanocrystals (e.g., Au, Ag, and Cu) hold great promise for driving photocatalytic reactions, but little is known about the plasmonic properties of Pd nanocrystals. Herein, we constructed a plasmonic Pd/Ru antenna–reactor photocatalyst through the controllable growth of a Ru nanoarray ‘reactor’ on a Pd nano-octahedron ‘antenna’ and demonstrated a plasmonic Pd-driven N₂ photofixation process. The plasmonic properties of Pd nano-octahedrons were verified using finite-difference time-domain (FDTD) simulations and refractive index sensitivity tests in water–glycerol mixtures. Notably, the constructed plasmonic antenna–reactor nanostructures exhibited superior photocatalytic activities during N₂ photofixation, with a maximum ammonia production rate of 117.5 ± 15.0 μmol g^{−1} h^{−1} under visible and near-infrared (NIR) light illumination. The mechanism can be attributed to the ability of the plasmonic Pd nanoantennas to harvest light to generate abundant hot electrons and the Ru nanoreactors to provide active sites for adsorption and activation of N₂. This work paves the way for the development of Pd-based plasmonic photocatalysts for efficient N₂ photofixation and sheds new light on the optimal design and construction of antenna–reactor nanostructures.

Received 5th June 2023

Accepted 4th September 2023

DOI: 10.1039/d3sc02862c

rsc.li/chemical-science

Introduction

Artificial ammonia (NH₃) synthesis from the reduction of nitrogen (N₂) is crucial for agricultural and industrial production, because NH₃ is an important building block for fertilizer and living forms.^{1–3} NH₃ has attracted great interest as a hydrogen storage material because of its high hydrogen capacity (17.6 wt%).⁴ Although there are abundant N₂ resources (approximately 78%) in the atmosphere, it is difficult to efficiently utilize N₂ because a quantity of energy is required to overcome the large bond energy (approximately 941 kJ mol^{−1}) of a stable N≡N triple bond.^{5,6} At present, the industrial production of NH₃ mainly adopts the classical Haber–Bosch process under high temperature and high pressure in the presence of highly pure N₂ and H₂, which is not only costly and requires the consumption of energy,^{7,8} but also leads to the emission of the greenhouse gas CO₂ for acquiring H₂ from the steam

reformation of natural gas.⁹ Alternatively, solar-driven chemical reactions represent a green and sustainable strategy to address the energy issues.^{10–12} Effective NH₃ synthesis could occur if a green and sustainable N₂ photofixation process could be developed with renewable solar energy and H₂O as the energy and hydrogen source, respectively.^{13–16} Among the various metal photocatalysts, ruthenium (Ru) with the appropriate N₂ adsorption energy is a satisfactory candidate for N₂ fixation, but its weak light-harvesting capability and lack of hot electrons to split the N≡N triple bond limit its applications in photocatalysis.^{17–19} Therefore, the design and construction of highly efficient Ru-based photocatalysts are promising but quite challenging.

Plasmonic metal nanocrystals (NCs) with localized surface plasmon resonance (LSPR) properties can generate highly energetic hot electrons to reduce the adsorbed molecules and effectively promote the conversion from light energy to chemical energy.^{20,21} Although substantial endeavours have been devoted to the exploration of Au, Ag, Cu, Al, and Fe as plasmonic metal photocatalysts, the LSPR properties of Pd NCs have rarely been reported.^{22–25} Interestingly, Pd not only provides sufficient active sites for molecular adsorption and activation, but it also harvests light as the plasmonic core to generate hot electrons through the manipulation of the spatial architecture.^{26–28} Consequently, the combination of a plasmonic Pd NC ‘antenna’ with a catalytically active Ru ‘reactor’ to fabricate antenna–reactor nanostructures offers the potential for driving N₂ photofixation.^{29–33} Specifically, Pd nano-octahedrons with large

^aCollege of Chemistry, Chemical Engineering and Materials Science, Shandong Normal University, Jinan 250014, China. E-mail: hljia@sdsu.edu.cn; cyzhang@sdsu.edu.cn

^bSchool of Chemistry and Chemical Engineering, Southeast University, Nanjing 211189, China. E-mail: zhangcy@seu.edu.cn

^cMinistry of Industry and Information Technology Key Lab of Micro-Nano Optoelectronic Information Systems and Guangdong Provincial Key Laboratory of Semiconductor Optoelectronic Materials and Intelligent Photonic Systems, Harbin Institute of Technology, Shenzhen 518055, China. E-mail: ruanqifeng@hit.edu.cn

† Electronic supplementary information (ESI) available. See DOI: <https://doi.org/10.1039/d3sc02862c>

‡ These authors contributed equally to this work.

cross-sections for light harvesting across a broad spectral range are good candidates as plasmonic nanoantennas.³⁴ Moreover, there has been great interest in Ru nanoarrays due to their larger surface area and a greater amount of active sites as compared to dense nanostructures.³⁵ Despite extensive efforts being devoted to the investigation of Au-, Ag-, and Cu-based antenna-reactor nanostructures in previous works, the fabrication of Pd-based antenna-reactor nanostructures remains a great challenge, but is extremely desirable.

In this work, we demonstrate a facile seed-mediated strategy for the controllable growth of Ru nanoarrays with different densities on a plasmonic Pd nano-octahedron core by precisely manipulating the growth kinetics. Theoretical and experimental studies were conducted to investigate the plasmonic properties of Pd nanoantennas. The plasmonic Pd/Ru antenna-reactor nanostructures exhibited excellent photocatalytic N₂ photofixation performance. In this nanostructure, plasmonic Pd nanoantennas harvest light to generate sufficient hot electrons, while the Ru nanoreactor provides active sites to adsorb/activate N₂ molecules and subsequently reduce it with hot electrons from the Pd antenna. In addition, the photocatalytic performance of the Pd/Ru antenna-reactor nanostructures during N₂ fixation was dependent upon the thickness of the Ru shells, with a maximum NH₃ production rate of $117.5 \pm 15.0 \mu\text{mol g}^{-1} \text{h}^{-1}$. The plasmon-driven N₂ fixation mechanism on the Pd/Ru antenna-reactor photocatalyst was systematically investigated as well.

Results and discussion

Fig. 1 illustrates the synthesis process for the preparation of Pd octahedron/Ru array nanostructures. Pd nano-octahedrons were prepared by employing Pd nanocubes as the seed.³⁶ For the overgrowth of Ru nanoarrays on Pd nano-octahedrons, ruthenium(III) acetylacetonate (Ru(acac)₃) was reduced by benzyl alcohol, with polyvinylpyrrolidone (PVP) as the surfactant. Because there is a lattice mismatch between Pd and Ru,³⁷ the deposition of Ru atoms on Pd nano-octahedrons prefers to occur in 3D island growth mode (Volmer–Weber mode), and Ru nanoarrays are subsequently formed on the Pd nano-octahedron surface.^{38,39} By changing the amount of the Ru precursor, the density and diameter of the Ru nanoarrays can be readily adjusted.

Pd nano-octahedrons with an average edge length of $43.4 \pm 4.7 \text{ nm}$ were employed as the seeds for the overgrowth of

Ru nanoarrays (Fig. 2a). The reasons for their utilization as the seeds are as follows: (1) the Pd nano-octahedrons possess an obvious plasmon peak that redshifts to the visible light region after the overgrowth of Ru, and therefore, they are a satisfactory candidate as a visible-light photocatalyst; (2) the finite-difference time-domain (FDTD) simulation results suggest that the extinction values are dominantly contributed by light absorption when the plasmon peak of Pd nano-octahedrons is less than 400 nm. As shown in the transmission electron microscopy (TEM) image, Ru nanoarrays with a thickness of $5.9 \pm 0.74 \text{ nm}$ have been successfully grown on the surface of Pd nano-octahedrons (Fig. 2b). The spatial architecture of Pd nano-octahedrons (Fig. 2c) and Pd/Ru bimetallic NCs (Fig. 2d) was further verified using scanning electron microscopy (SEM) images. The SEM results confirmed that Pd NCs possess an octahedral structure, and their size and morphology distributions are notably homogeneous. After the overgrowth of Ru nanoarrays, the octahedral morphology remained unchanged. The plasmonic properties of Pd NCs were revealed with extinction spectra, and the presence of an extinction peak at approximately 400 nm was attributed to the LSPR of Pd nano-octahedrons (Fig. 2e).³⁴ Because the LSPR wavelengths of plasmonic NCs are highly sensitive to the surrounding medium, an increase in the refractive index of the surrounding medium can induce a redshift of the plasmon resonance.⁴⁰ The plasmon peak exhibited a redshift from 403 nm to 502 nm after the Ru nanoarray overgrowth, which was caused by the larger refractive index of Ru as compared to water. To further examine the structure and composition of the Pd octahedron/Ru array nanostructures, high-angle annular dark-field scanning transmission electron microscopy (HAADF STEM) and corresponding energy-dispersive X-ray (EDX) element mapping were carried out (Fig. 2f). The EDX elemental mapping results clearly demonstrate that elemental Pd with an octahedral shape is located inside, while elemental Ru is coated outside. This is consistent with the observation from the elemental profile results (Fig. S1†). The representative high-resolution TEM (HRTEM) image of a single Pd octahedron/Ru array nanostructure suggests that Ru nanoarrays perpendicularly grow on the surface of Pd nano-octahedrons with an array interval of $1.41 \pm 0.3 \text{ nm}$ (Fig. 2g). As observed in the HRTEM image (Fig. 2h), lattice fringes of 2.239 \AA were assigned to face-centered cubic (fcc) Pd {111} facets, while lattice fringes of 2.204 \AA corresponded to the fcc Ru {111} facets. The lattice mismatch between Pd and Ru led to the 3D island epitaxial

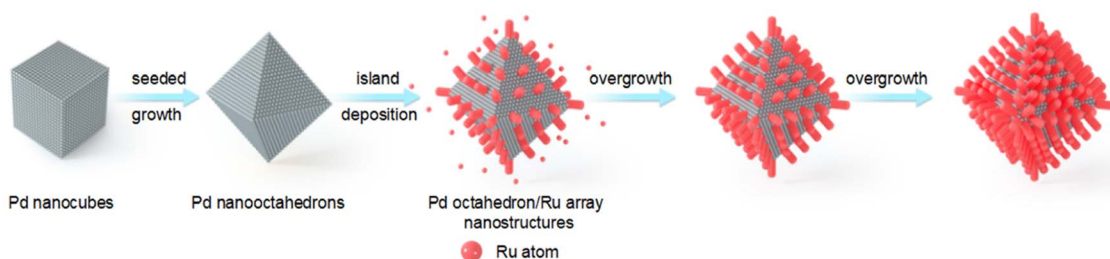


Fig. 1 Schematic illustrating the synthesis process of the Pd octahedron/Ru array nanostructures.



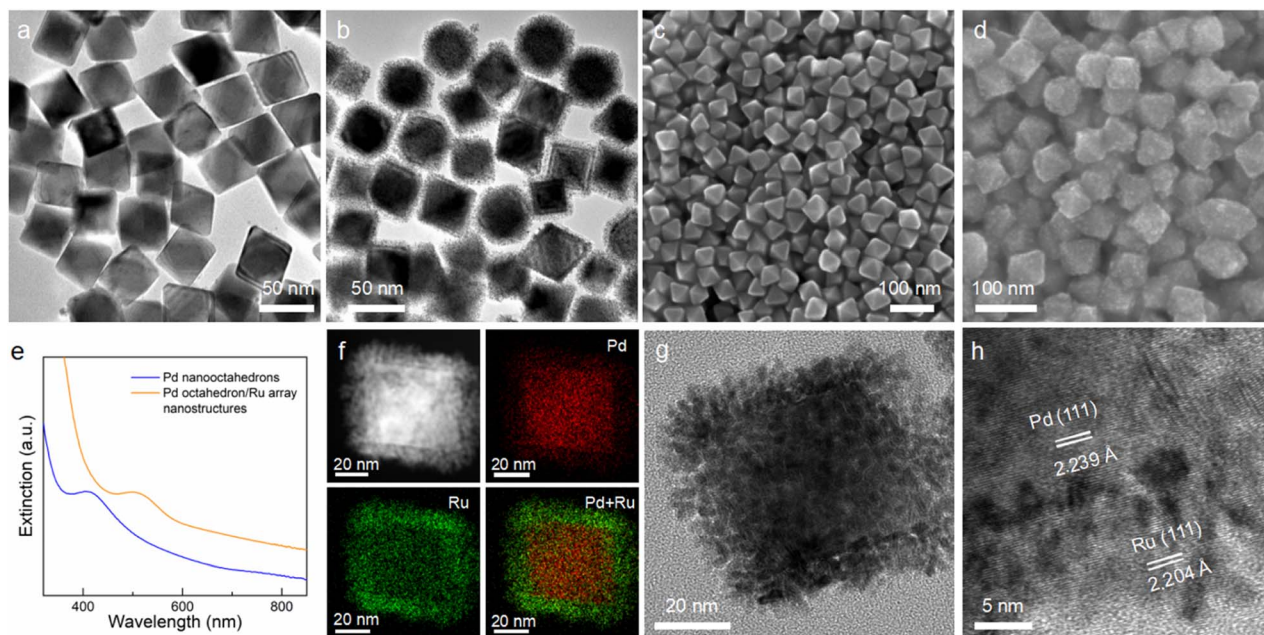


Fig. 2 Pd octahedron/Ru array nanostructures. TEM images of (a) Pd nano-octahedrons and (b) Pd octahedron/Ru array nanostructures. SEM images of (c) Pd nano-octahedrons and (d) Pd octahedron/Ru array nanostructures. (e) Extinction spectra of Pd nano-octahedrons (blue) and Pd octahedron/Ru array nanostructures (orange). (f) HAADF-STEM image (top left) and the corresponding EDX maps (top right and bottom) of a single Pd octahedron/Ru array nanostructure. (g) HRTEM image of a single Pd octahedron/Ru array nanostructure. (h) HRTEM image at the Pd–Ru interface.

growth of Ru nanoarrays.⁴¹ A Ru nanoarray shell with a spatially separated structure is more advantageous for charge separation than a dense shell, implying that it is an ideal photocatalyst for N_2 fixation.^{42,43}

The crystalline structure and chemical constituents of the Pd octahedron/Ru array nanostructures were further investigated with powder X-ray diffraction (PXRD) and X-ray photoelectron spectroscopy (XPS). There are only two groups of diffraction patterns that correspond to standard Pd and Ru peaks in the XRD spectrum (Fig. S2†), indicating that Pd and Ru exist in the metallic phase in Pd octahedron/Ru array nanostructures.⁴⁴ In addition, the representative XPS survey spectrum of the Pd octahedron/Ru array nanostructures is in accordance with the PXRD results. The high-resolution Pd 3d XPS spectra display two peaks after *in situ* Ar^+ ion sputtering for 40 s (Fig. S3b†), which were ascribed to the Pd 3d_{3/2} (340.5 eV) and Pd 3d_{5/2} (335.3 eV) of metallic Pd.⁴⁵ The Ru 3d XPS spectra were fitted with three peaks, including two Ru^0 peaks (Ru^0 3d_{3/2} at 285.4 eV and Ru^0 3d_{5/2} at 280.2 eV) and one Ru^{4+} peak (Fig. S3c†). The Ru^{4+} XPS peak originated from the inevitable occurrence of surface oxidation.⁴⁶ The Ru^{4+}/Ru^0 ratio was calculated to be 1 : 11, which confirms that elemental Ru exists primarily in the Ru^0 state. Moreover, the presence of $Ru\ 3p_{1/2}$ and $Ru\ 3p_{2/3}$ peaks further identifies the metallic Ru^0 state (Fig. S3d†).⁴⁷

To gain a more complete understanding of the LSPR properties of Pd nano-octahedrons, theoretical calculations and experimental measurements were conducted. First, we calculated the extinction, scattering, and absorption cross sections of Pd nano-octahedrons with edge lengths of 30–80 nm using

FDTD simulation (Fig. 3a–f). Fig. 3g shows the dependence of the extinction spectra on the edge length of the Pd nano-octahedrons. The simulated extinction cross-section increases and the longitudinal plasmon peak redshifts with increasing edge length of the Pd nano-octahedrons. A single LSPR peak can be clearly observed when the edge length is larger than approximately 40 nm, which is predominantly contributed by absorption rather than scattering (Fig. 3a–f).⁴⁸ Despite some discrepancies between the calculated and measured plasmon peaks, the simulation results revealed the plasmonic properties of the Pd nano-octahedrons. The difference was attributed to the nonuniform morphology and wide size distribution of the prepared Pd nano-octahedrons.⁴⁹ Next, the plasmonic properties of the Pd nano-octahedrons were further examined using a refractive index sensitivity test. Because the LSPR peaks are highly sensitive to the surrounding dielectric environment, the change in the refractive index of the surrounding environment will cause a shift in plasmon resonance peaks.⁵⁰ In this work, Pd nano-octahedrons were redispersed into a series of water–glycerol mixture solutions of varying volume ratios to detect the refractive index sensitivity.⁵¹ Fig. 3h displays the refractive index-dependence of the plasmon peaks in different mixture solutions. As the volume ratio of glycerol and the refractive index of the mixtures increased, the plasmon resonance peaks redshifted. Moreover, there is a linear dependence of the plasmon shift on the refractive index of the mixture (Fig. 3i). The slope of this line is 230 nm RIU^{-1} , which indicates the excellent refractive index sensitivity of the Pd octahedron nanoparticles.⁵² Notably, the plasmon peak of the Pd nano-octahedrons in



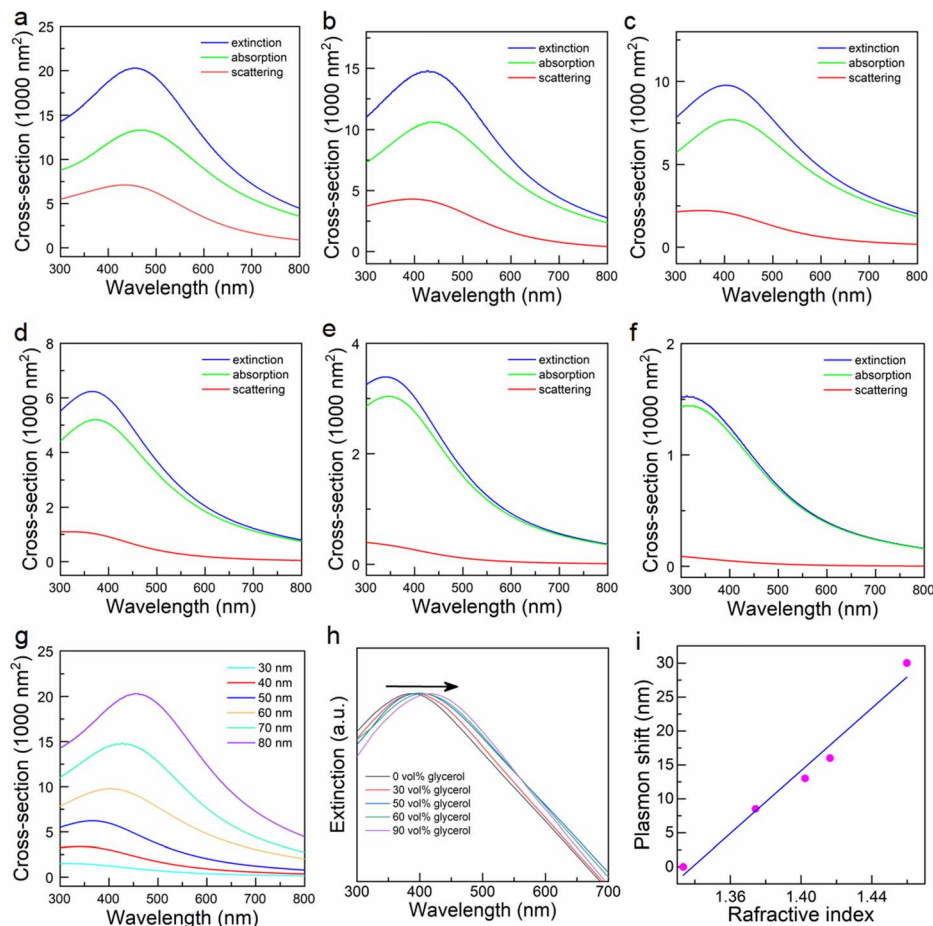


Fig. 3 Plasmonic properties of Pd nano-octahedrons. (a–f) Simulated extinction (blue), absorption (green), and scattering (red) spectra of Pd nano-octahedrons with an edge length of (a) 80 nm, (b) 70 nm, (c) 60 nm, (d) 50 nm, (e) 40 nm, and (f) 30 nm. (g) Comparison of the simulated extinction spectra of Pd nano-octahedrons with different edge lengths. (h) Normalized extinction spectra of Pd nano-octahedrons dispersed in water–glycerol mixtures of varying compositions. The arrow indicates the direction of increasing volume percentage of glycerol. (i) Dependence of the plasmon shift on the refractive index of the liquid mixture.

Fig. 3h is not consistent with the calculation result. The discrepancy may be ascribed to the difference in the size and morphology distribution between the theoretical calculation and the experimental results. The Pd nano-octahedrons used for the FDTD simulations were uniform in shape and size, but the yield of the obtained Pd nano-octahedrons was not 100% and the morphology was not perfectly octahedral, which led to a deviation between the theoretical and experimental results. To further confirm the LSPR properties of Pd nano-octahedrons, we performed surface photovoltage (SPV) spectroscopy. Under plasmon resonance excitation, the local electric field around the Pd nano-octahedrons was largely enhanced, inducing a surface voltage enhancement in the SPV spectrum. As shown in Fig. S4,[†] the SPV signal faithfully followed the surface plasmon resonance of the Pd nano-octahedrons, confirming the LSPR properties of the Pd nano-octahedrons. Above all, the theoretical and experimental results substantiated the plasmonic property of the Pd nano-octahedrons, suggesting that a Pd nano-octahedron sample is a satisfactory candidate as a nano-antenna for plasmon-driven photocatalysis. Pd nanocubes that

were seeds for the preparation of Pd nano-octahedrons were also employed to fabricate Pd cube/Ru nanostructures (Fig. S5[†]). There were no plasmon resonance peaks in the visible and near-infrared (NIR) regions for Pd nanocubes or Pd cube/Ru nanostructures (Fig. S5c and d[†]), which may be caused by the small size ($13.2 \pm 1.2 \text{ nm}$) of the Pd nanocubes. Therefore, Pd nano-octahedrons were selected as plasmonic antennas for visible-/NIR-light photocatalysis.

Precisely adjusting the thickness of the Ru nanoarray shell is important for its photocatalytic performance. The Pd octahedron/Ru array nanostructures with different shell thicknesses were obtained by increasing the concentration of the $\text{Ru}(\text{acac})_3$ precursor from 0.25 mg mL^{-1} to 2.5 mg mL^{-1} (Fig. 4 and S6[†]). The shell thickness displayed a linear dependence on the $\text{Ru}(\text{acac})_3$ concentration, with the minimum and maximum values being $3.4 \pm 0.4 \text{ nm}$ and $9.8 \pm 0.9 \text{ nm}$, respectively (Fig. 4i). Ru nanoarrays with sparse spatial distribution were observed when the $\text{Ru}(\text{acac})_3$ concentration was less than 0.75 mg mL^{-1} (Fig. S6a and b[†]). In addition, the gap distance in the Ru nanoarrays can be controlled in the range of 0.9–2.5 nm



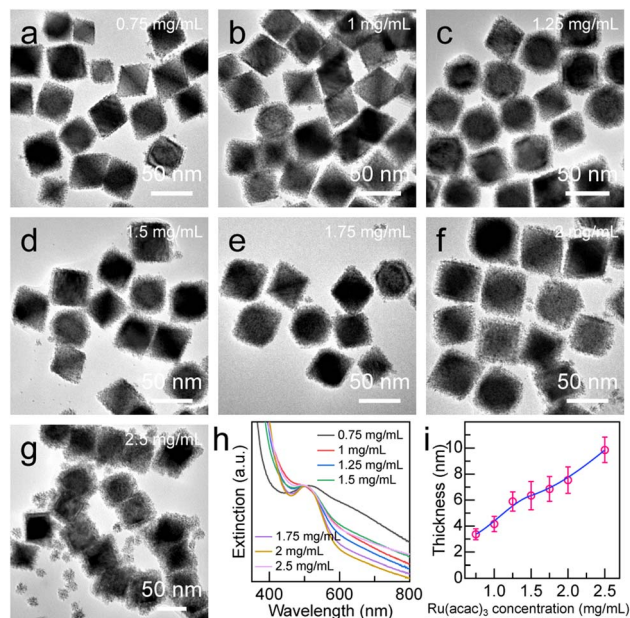


Fig. 4 Effect of precursor concentration on Ru shell thickness. (a–g) Representative TEM images of the Pd octahedron/Ru array nanostructures produced at different $\text{Ru}(\text{acac})_3$ concentrations. (h) Extinction spectra of the Pd octahedron/Ru array nanostructures obtained when different $\text{Ru}(\text{acac})_3$ concentrations were employed for Ru shell growth. (i) Ru shell thickness as a function of $\text{Ru}(\text{acac})_3$ concentration.

by simply adjusting the amount of Ru precursor (Fig. S6†). When the concentration of Ru precursor is equal to or greater than 2.5 mg mL^{-1} , small Ru nanoparticles are generated due to self-nucleation, which may be caused by the greater deposition rate rather than the diffusion rate (Fig. 4g).⁵³

A few control experiments were conducted to investigate the growth behaviour, including precursor injection rate (Fig. S7†), reaction temperature (Fig. S8†), PVP molecular weight (Fig. S9†), amount of added PVP (Fig. S10†), and Ru precursor species (Fig. S11†). First, the precursor injection rate can affect the growth behaviour.^{54,55} When the injection rate of the $\text{Ru}(\text{acac})_3$ precursor was 2 mL h^{-1} , a large amount of self-nucleated Ru nanoparticles was formed due to the rapid reduction of Ru precursor by benzyl alcohol (Fig. S7a†). When the injection rate of the $\text{Ru}(\text{acac})_3$ precursor was greater than 8 mL h^{-1} , the Ru nanoparticles tended to grow on Pd seeds, and Pd octahedron/Ru array nanostructures were obtained (Fig. S7b and c†). Thus, a relatively moderate precursor injection rate is of great importance for the formation of Ru array nanostructures. Next, we investigated the effect of reaction temperature on Ru deposition. A layer of ultrathin Ru shells was deposited on the side edges of Pd nano-octahedrons when the reaction temperature was changed from 180°C to 140°C due to the lower reduction rate of the Ru precursor (Fig. S8b†). However, no Ru nanoarrays were observed once the reaction temperature was decreased to 100°C due to the poor reduction capability of benzyl alcohol at this temperature (Fig. S8a†).⁵⁶ Third, we investigated the effect of the molecular weight and the amount of PVP molecules added on the growth behaviour. A relatively

large molecular weight and an appropriate addition of PVP are of great importance to the overgrowth of Ru nanoarrays on Pd nano-octahedrons (Fig. S9 and S10†). Fourth, when the $\text{Ru}(\text{acac})_3$ precursor was replaced with RuCl_3 , the products of the Pd octahedron/Ru array nanostructures were obtained, but the plasmon peak vanished (Fig. S11†). Finally, hollow Ru nanoshells were obtained by etching the Pd core from the Pd octahedron/Ru array nanostructures (Fig. S12†), indicating the excellent stability of Ru shells.⁵⁷

The Pd octahedron/Ru array nanostructures with plasmonic Pd antennas for harvesting light and catalytically active Ru reactors for N_2 adsorption/activation are satisfactory candidates as plasmonic antenna-reactor catalysts for N_2 photofixation. To explore the photocatalytic performance of the Pd octahedron/Ru array nanostructures, N_2 reduction experiments were conducted under visible and NIR light illumination ($\lambda > 400 \text{ nm}$). In this study, methanol was selected as the hole scavenger. To investigate the dependence of N_2 photofixation activity on Ru array thickness, six types of Pd octahedron/Ru array nanostructures with different array thicknesses ($3.4 \pm 0.4 \text{ nm}$ (Fig. 4a), $4.2 \pm 0.6 \text{ nm}$ (Fig. 4b), $5.9 \pm 0.7 \text{ nm}$ (Fig. 4c), $6.4 \pm 1.1 \text{ nm}$ (Fig. 4d), $6.9 \pm 1.3 \text{ nm}$ (Fig. 4e), and $7.5 \pm 1.0 \text{ nm}$ (Fig. 4f)) were synthesized and employed as photocatalysts, named Pd/Ru_{3.4}, Pd/Ru_{4.2}, Pd/Ru_{5.9}, Pd/Ru_{6.4}, Pd/Ru_{6.9}, and Pd/Ru_{7.5}. For comparison, Pd nano-octahedrons (Fig. 2a), Ru NCs (Fig. S13†), Pd cube/Ru nanostructures (Fig. S5b†), and the mixture of Pd nano-octahedrons with Ru NCs were prepared as the photocatalysts. Notably, although most of the surfactants were removed by centrifugation, some residual PVP molecules remained on the catalyst surface to maintain the stability of the colloidal nanoparticles, which may affect the catalytic activity. Although the surfactant molecules can block the active sites and decrease the catalytic activity, the presence of some molecules on catalysts can enhance the catalytic activity and selectivity through the electron donation effect or geometric changes.^{58–60} In this work, PVP was employed as the surfactant for all the catalysts. Its influence on catalytic activity should be similar and was not taken into account. The catalyst amount and the Pd/Ru molar ratio in the Pd/Ru catalyst were determined by inductively coupled plasma-optical emission spectroscopy (ICP-OES) (Fig. S14 and Table S1†). The amount of NH_3 production was quantitatively detected by the indophenol blue method.⁶¹ The linear calibration relationship between the standard NH_4^+ concentrations and the corresponding absorbance peak values is displayed in Fig. S15.†

As shown in Fig. 5a, the individual Pd nano-octahedrons or Ru NCs exhibit weak N_2 photofixation activity due to the lack of catalysts for N_2 adsorption or the weak light-harvesting capability. The photocatalytic N_2 fixation activity on the Pd cube/Ru nanostructures was also weak because there was an absence of plasmonic properties for the Pd nanocubes. In contrast, the NH_3 production rate for the Pd/Ru_{5.9} sample was 3.6-fold higher than that of the mixture sample after 2 h of illumination, indicating that the close coupling between the plasmonic antenna and active reactor is crucial for the antenna-reactor photocatalyst. Pd nano-octahedrons or Ru NCs independently contribute to the photocatalytic performance of the mixture



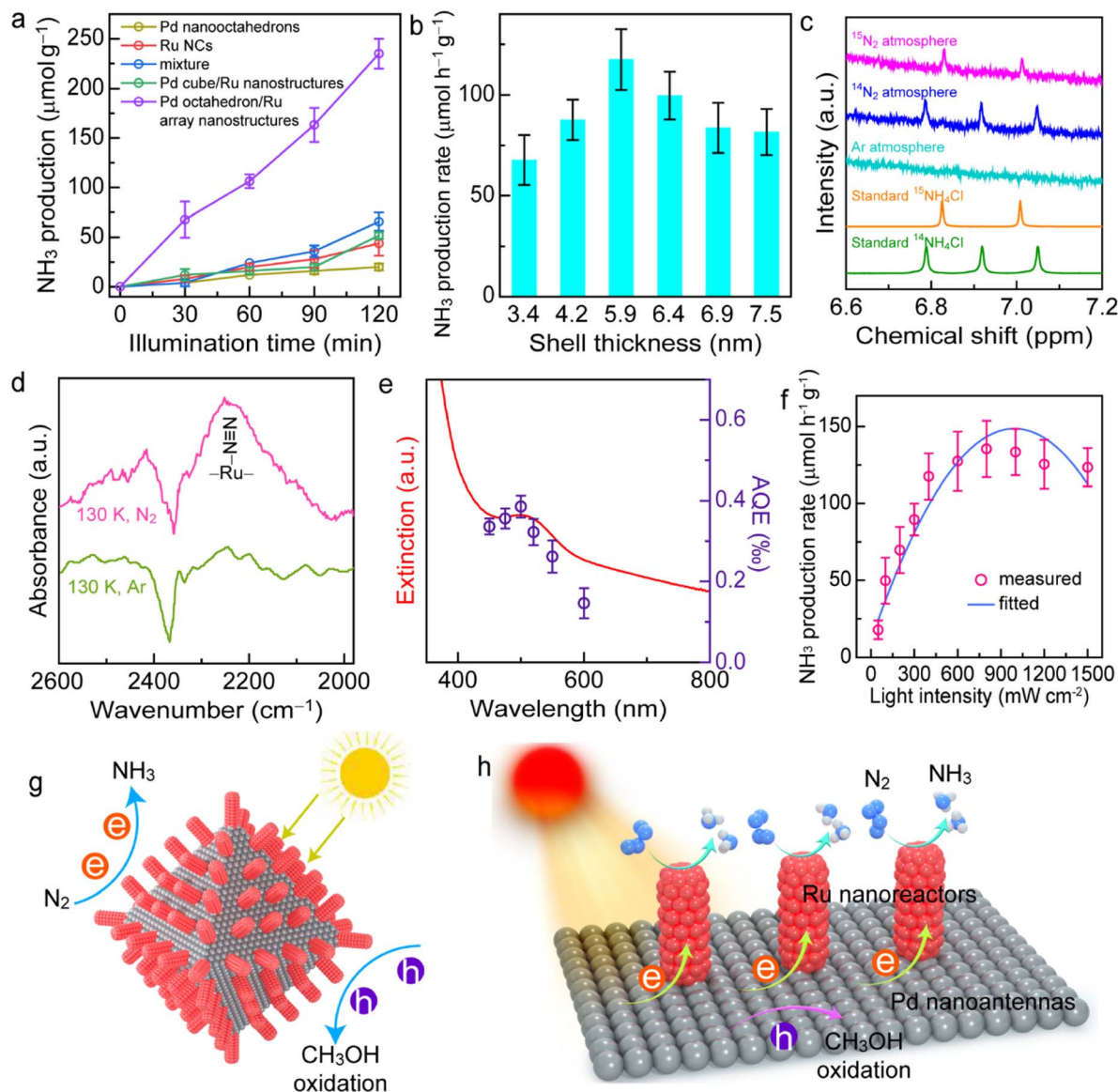


Fig. 5 Plasmon-driven N₂ photofixation. (a) NH₃ production amounts as functions of illumination time using different photocatalysts. (b) Dependence of the N₂ photofixation activity upon the Ru shell thickness. (c) ¹H NMR spectra of the reaction solutions under Ar, ¹⁴N₂, and ¹⁵N₂ atmospheres. (d) LT-FTIR spectra of the Pd/Ru_{5.9} sample under N₂ and Ar atmospheres. (e) Extinction spectrum (left axis) and wavelength-dependent apparent quantum efficiency of NH₃ evolution (right axis) of the Pd/Ru_{5.9} sample. (f) Dependence of the photocatalytic NH₃ production rate upon the light intensity. (g) Schematics illustrating N₂ photofixation on Pd octahedron/Ru array nanostructures. (h) Mechanism of N₂ photofixation on the Pd octahedron/Ru array nanostructure. e, hot electron; h, hot hole.

sample. The N₂ photofixation activity exhibited a volcano-shaped dependence on the array thickness of the Pd octahedron/Ru array nanostructures (Fig. 5b), with a maximum NH₃ production rate of $117.5 \pm 15.0 \mu\text{mol g}^{-1} \text{h}^{-1}$. A thin Ru shell leads to weak photocatalytic activity due to insufficient Ru reactors on the Pd nanoantennas. Contrarily, a Ru shell that is too thick with high-density Ru nanoarrays can cover the active sites on the plasmonic Pd nanoantennas, which hinders the transfer of hot holes and the entrance of hole scavengers, eventually resulting in charge carrier recombination and a decrease in photocatalytic performance. It was found that an appropriate Ru shell thickness of $5.9 \pm 0.7 \text{ nm}$ exhibits the most optimal photocatalytic N₂ fixation activity. Notably, although

the indophenol blue method has been widely applied for the quantitative detection of ammonia concentration, interference may occur due to many factors such as pH value, organic compounds, and impurities. The results will be more convincing if a cross-check of the ammonia concentration can be performed using two different quantitative methods.^{62–64} To confirm the N₂ fixation activity, the generated ammonia concentrations were analyzed using an ion chromatography method (Fig. S16†). The NH₃ production rates determined by the ion chromatography method were consistent with those obtained by the indophenol blue method. In addition, we confirmed the N₂ fixation rates with a quantitative isotopically labelled ¹H nuclear magnetic resonance (NMR) method

(Fig. S17†). The N_2 photofixation experiment was conducted for 6 h with 3 mg of catalyst, and ammonia concentrations were determined by an indophenol blue method and an NMR method. The values of the produced NH_3 determined by the indophenol blue method were consistent with those obtained by the NMR method. Therefore, ion chromatography and NMR methods confirmed the N_2 fixation activity of the catalysts.

To investigate the origin of produced NH_3 , a series of control experiments under different conditions was carried out (Fig. S18†). No NH_3 can be detected in the absence of catalysts, light irradiation, or N_2 atmosphere, which suggests that NH_3 production occurred from photocatalytic N_2 fixation. The photocatalytic activity of the Pd octahedron/Ru array nanostructures decreased to $29.7 \mu\text{mol g}^{-1} \text{h}^{-1}$ in the absence of a sacrificial agent, revealing that the addition of hole scavengers increased the charge separation efficiency.⁶¹ To investigate the effect of hole sacrificial reagents, we employed four types of sacrificial reagents: ethanol, triethanolamine (TEOA), triethylamine (TEA), and Na_2SO_3 . Notably, the presence of TEA in the reaction system led to the aggregation of catalyst. Among the different types of hole scavengers, CH_3OH exhibited the strongest N_2 fixation performance (Fig. S19†). Thus, CH_3OH was selected as the hole sacrificial reagent in this work. To further confirm the nitrogen source for NH_3 production, we performed photocatalytic N_2 reduction experiments under Ar, $^{14}N_2$, and $^{15}N_2$ atmospheres and analyzed the reaction solution using 1H NMR spectroscopy (Fig. 5c and S20†). As displayed in Fig. 5c, it is difficult to observe $^{14}NH_4^+$ under an Ar atmosphere, suggesting that the reduction of possible $^{14}N_2$ in our system exerted no apparent influence on the NMR peaks. The double peaks associated with $^{15}NH_4^+$ were clearly observed in the 1H NMR spectrum under an $^{15}N_2$ atmosphere, while three peaks attributed to $^{14}NH_4^+$ were detected under an $^{14}N_2$ -atmosphere, which suggests that the nitrogen source for NH_3 -production was from N_2 . In addition, the byproduct N_2H_4 was not detected in the typical photocatalytic reaction using a spectrophotometric method, suggesting the high selectivity of photocatalytic N_2 fixation (Fig. S21†).⁶⁵ H_2 or CO_2 may also be generated through either the reduction of water by hot electrons or the oxidation of methanol by hot holes. To investigate whether there is H_2 or CO_2 evolution in this reaction process, we conducted N_2 photofixation experiments in a gastight glass reactor. No H_2 was detected by gas chromatography after a 2 h reaction (Fig. S22a†), suggesting that H_2 evolution is unfavourable under an N_2 atmosphere. However, CO_2 was detected in this photocatalytic system, and the amount of CO_2 increased over time (Fig. S22b†). The evolution of CO_2 originated from the oxidation half-reaction driven by hot holes, which further confirmed that NH_3 was from the reduction half-reaction driven by hot electrons. Because an off-line gas chromatograph was employed for the detection of gaseous products, the air may affect this detection. Thus, a CO_2 signal was detected at 0 h. To determine the level of recyclability of the Pd octahedron/Ru array nanostructures, the photocatalytic reaction was conducted in three successive cycles. Although there was a 10.6% decrease in the photocatalytic activity after three cycles, the catalyst maintained satisfactory recyclability (Fig. S23†). The

decrease in the photocatalytic activity may be caused by the loss of catalyst during centrifugation. Moreover, the stability of the Pd octahedron/Ru array nanostructures was examined by performing TEM imaging, and XRD and XPS measurement of the catalyst after 2 h of N_2 photofixation (Fig. S24†). The morphology, crystalline nature, and chemical states of the catalyst remained unchanged, confirming the excellent stability of the photocatalysts.

The excellent photocatalytic performance of the Pd octahedron/Ru array nanostructures was mainly contributed by the antenna-reactor mechanism. First, we investigated the active sites for N_2 adsorption using theoretical and experimental studies. Density functional theory (DFT) calculations showed that the adsorption energy of N_2 on Ru (-0.534 eV) was more negative than that on Pd (0.145 eV) (Fig. S25†), which revealed that the activation sites for N_2 fixation were on the Ru nanoarrays. To gain a deeper insight into the N_2 adsorption, we conducted N_2 temperature-programmed desorption (TPD) (Fig. S26†) and low-temperature Fourier-transform infrared (LT-FTIR) (Fig. 5d) spectroscopy measurements. Two peaks appeared in the N_2 TPD spectrum, with one peak located at 334°C that was attributed to physisorption and the other peak at 498°C that originated from N_2 chemisorption, revealing the presence of active sites on the Pd octahedron/Ru array nanostructures for N_2 adsorption. In addition, a vibration peak located at 2252 cm^{-1} in the LT-FTIR spectra was detected under an N_2 atmosphere, suggesting that N_2 molecules adsorb on the Ru catalyst *via* an end-on configuration.¹⁸ Next, we examined the plasmon-driven N_2 photofixation process. To reveal the contribution of the photothermal effect upon the N_2 fixation, we carried out N_2 reduction experiments in water baths set at different temperatures in the dark (Fig. S27†). The results showed that the effect of photothermal heating on the photocatalytic activity was small, although it can hardly be ruled out. To further verify the plasmon-driven N_2 fixation process, we calculated the apparent quantum efficiency (AQE) by performing N_2 fixation experiments under different levels of monochromatic light illumination, and the action spectrum was therefore acquired (Fig. 5e). The trend of AQE values was in accordance with the absorption spectrum, which definitely confirmed that the photocatalytic N_2 fixation was driven by the plasmon resonance of the Pd nano-octahedrons.⁶¹ In addition, there was a nearly linear dependence of the NH_3 production rate on the Pd octahedron/Ru array nanostructures when the light intensity was less than 800 mW cm^{-2} (Fig. 5f). Further increase of the light intensity caused a decrease in photocatalytic activity, with the maximum NH_3 production rate at 800 mW cm^{-2} . The linear relationship between activity and light intensity suggested that the N_2 photofixation was driven by plasmonic hot electrons. On the basis of the above results, we proposed a possible mechanism for N_2 photofixation on Pd octahedron/Ru array nanostructures (Fig. 5g and h). In the nanostructures, Ru reactors afford active sites to adsorb and activate N_2 -molecules, while Pd antennas interact with light and generate hot electrons for N_2 reduction. The hot holes are consumed using hole scavengers (CH_3OH) to close the photocatalytic cycle.



The Ru array nanostructures also facilitated charge separation and promoted the photocatalytic activity.

Conclusions

We constructed a novel Pd-based antenna-reactor plasmonic photocatalyst through the overgrowth of Ru nanoarrays on Pd nano-octahedrons. The plasmonic properties of the Pd nano-antennas were systematically investigated using theoretical and experimental studies. The obtained Pd octahedron/Ru array nanostructures exhibited an excellent photocatalytic N₂ fixation performance. The mechanism may be attributed to the fact that Ru reactors adsorb and activate N₂ molecules and that Pd octahedron antennas capture light and effectively generate energetic hot electrons for N₂ reduction. This work provides a new strategy for the construction of plasmonic Pd-based antenna-reactor nanostructures for photocatalysis and opens a new avenue to explore potential plasmonic metal photocatalysts.

Data availability

The data that support the findings of this study are available in the main text and the ESI.†

Author contributions

Y. Y. Yang and H. L. Jia contributed equally to this work. H. L. Jia and C.-y. Zhang conceived and supervised the project, and wrote the paper. Y. Y. Yang performed the experiments. S. H. Su and Q. F. Ruan carried out the FDTD simulations and contributed to the data interpretation. Y. D. Zhang, M. X. Zhao, and J. Z. Li assisted with the photocatalytic experiments and participated in discussions. All authors discussed the results and commented on the manuscript.

Conflicts of interest

There are no conflicts to declare.

Acknowledgements

This work was supported by the National Natural Science Foundation of China (Grant No. 21735003), the Natural Science Foundation of Shandong Province (No. ZR2020MB040), and the Natural Science Foundation of Guangdong Province (No. 2023A1515012912).

Notes and references

- 1 B. H. R. Suryanto, K. Matuszek, J. Choi, R. Y. Hodgetts, H.-L. Du, J. M. Bakker, C. S. M. Kang, P. V. Cherepanov, A. N. Simonov and D. R. MacFarlane, *Science*, 2021, **372**, 1187–1191.
- 2 Y. Ashida, K. Arashiba, K. Nakajima and Y. Nishibayashi, *Nature*, 2019, **568**, 536–540.
- 3 L. C. Seefeldt, Z.-Y. Yang, D. A. Lukoyanov, D. F. Harris, D. R. Dean, S. Raugei and B. H. Hoffman, *Chem. Rev.*, 2020, **120**, 5082–5106.
- 4 J. P. Guo and P. Chen, *Chem*, 2017, **3**, 709–712.
- 5 J. H. Yang, Y. Z. Guo, W. Z. Lu, R. B. Jiang and J. F. Wang, *Adv. Mater.*, 2018, **30**, 1802227.
- 6 J. Y. Li, R. M. Chen, J. L. Wang, Y. Zhou, G. D. Yang and F. Dong, *Nat. Commun.*, 2022, **13**, 1098.
- 7 S. S. Shang, W. Xiong, C. Yang, B. Johannessen, R. G. Liu, H.-Y. Hsu, Q. F. Gu, M. K. H. Leung and J. Shang, *ACS Nano*, 2021, **15**, 9670–9678.
- 8 W. S. Huang, L.-Y. Peng, J. Y. Zhang, C. R. Liu, G. Y. Song, J.-H. Su, W.-H. Fang, G. L. Cui and S. W. Hu, *J. Am. Chem. Soc.*, 2023, **145**, 811–821.
- 9 S.-L. Meng, X.-B. Li, C.-H. Tung and L.-Z. Wu, *Chem*, 2021, **7**, 1431–1450.
- 10 H. W. Lin, S. Q. Luo, H. B. Zhang and J. H. Ye, *Joule*, 2022, **6**, 294–314.
- 11 W. H. Huang, C. Y. Su, C. Zhu, T. T. Bo, S. W. Zuo, W. Zhou, Y. F. Ren, Y. N. Zhang, J. Zhang, M. Rueping and H. B. Zhang, *Angew. Chem., Int. Ed.*, 2023, **62**, e202304634.
- 12 W. H. Huang, T. T. Bo, S. W. Zuo, Y. Z. Wang, J. M. Chen, S. Ould-Chikh, Y. Li, W. Zhou, J. Zhang and H. B. Zhang, *SusMat*, 2022, **2**, 466–475.
- 13 H. Hirakawa, M. Hashimoto, Y. Shiraishi and T. Hirai, *J. Am. Chem. Soc.*, 2017, **139**, 10929–10936.
- 14 N. Zhang, A. Jalil, D. X. Wu, S. M. Chen, Y. F. Liu, C. Gao, W. Ye, Z. M. Qi, H. X. Ju, C. M. Wang, X. J. Wu, L. Song, J. F. Zhu and Y. J. Xiong, *J. Am. Chem. Soc.*, 2018, **140**, 9434–9443.
- 15 H. L. Jia, A. X. Du, H. Zhang, J. H. Yang, R. B. Jiang, J. F. Wang and C.-y. Zhang, *J. Am. Chem. Soc.*, 2019, **141**, 5083–5086.
- 16 H. L. Jia, M. X. Zhao, A. X. Du, Y. R. Dou and C.-y. Zhang, *Chem. Sci.*, 2022, **13**, 13060–13067.
- 17 M. Kitano, Y. Inoue, Y. Yamazaki, F. Hayashi, S. Kanbara, S. Matsuishi, T. Yokoyama, S.-W. Kim, M. Hara and H. Hosono, *Nat. Chem.*, 2012, **4**, 934–940.
- 18 C. Y. Hu, X. Chen, J. B. Jin, Y. Han, S. M. Chen, H. X. Ju, J. Cai, Y. R. Qiu, C. Gao, C. M. Wang, Z. M. Qi, R. Long, L. Song, Z. Liu and Y. J. Xiong, *J. Am. Chem. Soc.*, 2019, **141**, 7807–7814.
- 19 H. L. Jia, Y. Y. Yang, Y. R. Dou, F. Li, M. X. Zhao and C.-y. Zhang, *Chem. Commun.*, 2022, **58**, 1013–1016.
- 20 S. Linic, P. Christopher and D. B. Ingram, *Nat. Mater.*, 2011, **10**, 911–921.
- 21 S. S. Li, H. Huang, L. Shao and J. F. Wang, *ACS Nano*, 2021, **15**, 10759–10768.
- 22 H. L. Jia, F. Li, T. H. Chow, X. Y. Liu, H. Zhang, Y. Lu, J. F. Wang and C.-y. Zhang, *Nano Lett.*, 2022, **22**, 7268–7274.
- 23 R. C. Elias and S. Linic, *J. Am. Chem. Soc.*, 2022, **144**, 19990–19998.
- 24 L. T. Quynh, C.-W. Cheng, C.-T. Huang, S. S. Raja, R. Mishra, M.-J. Yu, Y.-J. Lu and S. Gwo, *ACS Nano*, 2022, **16**, 5975–5983.
- 25 H. B. Zhang, T. Wang, J. J. Wang, H. M. Liu, T. D. Dao, M. Li, G. G. Liu, X. G. Meng, K. Chang, L. Shi, T. Nagao and J. H. Ye, *Adv. Mater.*, 2016, **28**, 3703–3710.



- 26 J. Guo, Y. Zhang, L. Shi, Y. F. Zhu, M. F. Mideksa, K. Hou, W. S. Zhao, D. W. Wang, M. T. Zhao, X. F. Zhang, J. W. Lv, J. Q. Zhang, X. L. Wang and Z. Y. Tang, *J. Am. Chem. Soc.*, 2017, **139**, 17964–17972.
- 27 C. Y. Hu, K. Q. Lin, X. L. Wang, S. J. Liu, J. Yi, Y. Tian, B. H. Wu, G. X. Chen, H. Y. Yang, Y. Dai, H. Li and N. F. Zheng, *J. Am. Chem. Soc.*, 2014, **136**, 12856–12859.
- 28 F. A. A. Nugroho, P. Bai, I. Darmadi, G. W. Castellanos, J. Fritzsche, C. Langhammer, J. G. Rivas and A. Baldi, *Nat. Commun.*, 2022, **13**, 5737.
- 29 D. F. Swearer, R. K. Leary, R. Newell, S. Yazdi, H. Robatjazi, Y. Zhang, D. Renard, P. Nordlander, P. A. Midgley, N. J. Halas and E. Ringe, *ACS Nano*, 2017, **11**, 10281–10288.
- 30 Y. G. Yuan, L. N. Zhou, H. Robatjazi, J. W. L. Bao, J. Y. Zhou, A. Bayles, L. Yuan, M. H. Lou, M. H. Lou, S. Khatiwada, E. A. Carter, P. Nordlander and N. J. Halas, *Science*, 2022, **378**, 889–893.
- 31 L. Yuan, J. Y. Zhou, M. Zhang, X. L. Wen, J. M. P. Martinez, H. Robatjazi, L. N. Zhou, E. A. Carter, P. Nordlander and N. J. Halas, *ACS Nano*, 2022, **16**, 17365–17375.
- 32 K. Sytwu, M. Vadai, F. Hayee, D. K. Angell, A. Dai, J. Dixon and J. A. Dionne, *Science*, 2021, **371**, 280–283.
- 33 H. L. Jia, F. Li, Y. Y. Yang, M. X. Zhao, J. Z. Li and C.-y. Zhang, *Chem. Sci.*, 2023, **14**, 5656–5664.
- 34 R. Long, Z. L. Rao, K. K. Mao, Y. Li, C. Zhang, Q. L. Liu, C. M. Wang, Z.-Y. Li, X. J. Wu and Y. J. Xiong, *Angew. Chem., Int. Ed.*, 2015, **54**, 2425–2430.
- 35 L. Q. Tian, Q. Xin, C. Zhao, G. C. Xie, M. Z. Akram, W. R. Wang, R. P. Ma, X. R. Jia, B. D. Guo and J. R. Gong, *Small*, 2021, **17**, 2006530.
- 36 M. S. Jin, H. Zhang, Z. X. Xie and Y. N. Xia, *Energy Environ. Sci.*, 2012, **5**, 6352–6357.
- 37 Y. C. Yao, D. S. He, Y. Lin, X. Q. Feng, X. Wang, P. Q. Yin, X. Hong, G. Zhou, Y. Wu and Y. D. Li, *Angew. Chem., Int. Ed.*, 2016, **55**, 5501–5505.
- 38 J. Liu and J. T. Zhang, *Chem. Rev.*, 2020, **120**, 2123–2170.
- 39 H. L. Jia, Y. Y. Yang, T. H. Chow, H. Zhang, X. Y. Liu, J. F. Wang and C.-y. Zhang, *Adv. Funct. Mater.*, 2021, **31**, 2101255.
- 40 H. J. Chen, L. Shao, Q. Li and J. F. Wang, *Chem. Soc. Rev.*, 2013, **42**, 2679–2724.
- 41 A. Janssen, Z. H. Lyu, M. Figueras-Valls, H.-Y. Chao, Y. F. Shi, V. Pawlik, M. F. Chi, M. Mavrikakis and Y. N. Xia, *Nano Lett.*, 2022, **22**, 3591–3597.
- 42 Z. B. Luo, C. C. Li, S. S. Liu, T. Wang and J. L. Gong, *Chem. Sci.*, 2017, **8**, 91–100.
- 43 C. W. Liu, Z. L. Qiu, W. L. Meng, J. W. Chen, J. J. Qi, C. Dong and M. T. Wang, *Nano Energy*, 2015, **12**, 59–68.
- 44 K. Kusada, H. Kobayashi, R. Ikeda, Y. Kubota, M. Takata, S. Toh, T. Yamamoto, S. Matsumura, N. Sumi, K. Sato, K. Nagaoka and H. Kitagawa, *J. Am. Chem. Soc.*, 2014, **136**, 1864–1871.
- 45 Z. C. Zhang, Y. Liu, B. Chen, Y. Gong, L. Gu, Z. X. Fan, N. L. Yang, Z. C. Lai, Y. Chen, J. Wang, Y. Huang, M. Sindoro, W. X. Niu, B. Li, Y. Zong, Y. H. Yang, X. Huang, F. W. Huo, W. Huang and H. Zhang, *Adv. Mater.*, 2016, **28**, 10282–10286.
- 46 H. Liu, W. W. Zheng, Y. Zhao and Y. Z. Zhou, *Anal. Chem.*, 2021, **93**, 4944–4951.
- 47 T. L. Jiang, K. Li, S. Park, K. Zheng, Y. H. Meng, Y. Yuan, Z. C. Liu, Z. X. Zhu, X. H. Zheng, S. Liu and W. Chen, *Nano Lett.*, 2022, **22**, 1741–1749.
- 48 H. L. Jia, C. H. Fang, X.-M. Zhu, Q. F. Ruan, Y.-X. J. Wang and J. F. Wang, *Langmuir*, 2015, **31**, 7418–7426.
- 49 W. H. Ni, X. S. Kou, Z. Yang and J. F. Wang, *ACS Nano*, 2008, **2**, 677–686.
- 50 J. P. Zheng, X. Z. Cheng, H. Zhang, X. P. Bai, R. Q. Ai, L. Shao and J. F. Wang, *Chem. Rev.*, 2021, **121**, 13342–13453.
- 51 H. J. Chen, L. Shao, K. C. Woo, T. Ming, H.-Q. Lin and J. F. Wang, *J. Phys. Chem. C*, 2009, **113**, 17691–17697.
- 52 R. Bukasov and J. S. Shumaker-Parry, *Nano Lett.*, 2007, **7**, 1113–1118.
- 53 W. C. Wang, X. Li, T. O. He, Y. M. Liu and M. S. Jin, *Nano Lett.*, 2019, **19**, 1743–1748.
- 54 S. F. Xie, S.-I. Choi, N. Lu, L. T. Roling, J. A. Herron, L. Zhang, J. Park, J. G. Wang, M. J. Kim, Z. X. Xie, M. Mavrikakis and Y. N. Xia, *Nano Lett.*, 2014, **14**, 3570–3576.
- 55 Y. N. Xia, K. D. Gilroy, H.-C. Peng and X. H. Xia, *Angew. Chem., Int. Ed.*, 2016, **55**, 2–38.
- 56 S. F. Xie, H.-C. Peng, N. Lu, J. G. Wang, M. J. Kim, Z. X. Xie and Y. N. Xia, *J. Am. Chem. Soc.*, 2013, **135**, 16658–16667.
- 57 H. H. Ye, Q. X. Wang, M. Catalano, N. Lu, J. Vermeylen, M. J. Kim, Y. Z. Liu, Y. G. Sun and X. H. Xia, *Nano Lett.*, 2016, **16**, 2812–2817.
- 58 G. X. Chen, C. F. Xu, X. Q. Huang, J. Y. Ye, L. Gu, G. Li, Z. C. Tang, B. H. Wu, H. Y. Yang, Z. P. Zhao, Z. Y. Zhou, G. Fu and N. F. Zheng, *Nat. Mater.*, 2016, **15**, 564–569.
- 59 Z. Cao, D. Kim, D. C. Hong, Y. Yu, J. Xu, S. Lin, X. D. Wen, E. M. Nichols, K. Jeong, J. A. Reimer, P. D. Yang and C. J. Chang, *J. Am. Chem. Soc.*, 2016, **138**, 8120–8125.
- 60 J. H. Hu, H. Mi, N. Wang, H. Y. Zhu, W. Y. Guo, S. R. Zhang, F. Shi, Z. B. Lei, Z.-H. Liu and R. B. Jiang, *Nanoscale*, 2018, **10**, 5607–5616.
- 61 J. H. Yang, Y. Z. Guo, R. B. Jiang, F. Qin, H. Zhang, W. Z. Lu, J. F. Wang and J. C. Yu, *J. Am. Chem. Soc.*, 2018, **140**, 8497–8508.
- 62 S. Z. Andersen, V. Čolić, S. Yang, J. A. Schwalbe, A. C. Nielander, J. M. McEnaney, K. Enemark–Rasmussen, J. G. Baker, A. R. Singh, B. A. Rohr, M. J. Statt, S. J. Blair, S. Mezzavilla, J. Kibsgaard, P. C. K. Vesborg, M. Cargnello, S. F. Bent, T. F. Jaramillo, I. E. L. Stephens, J. K. Nørskov and I. Chorkendorff, *Nature*, 2019, **570**, 504–508.
- 63 C. Tang and S.-Z. Qiao, *Chem. Soc. Rev.*, 2019, **48**, 3166–3180.
- 64 Y. X. Zhao, F. Wu, Y. X. Miao, C. Zhou, N. Xu, R. Shi, L.-Z. Wu, J. W. Tang and T. R. Zhang, *Angew. Chem., Int. Ed.*, 2021, **60**, 21728–21731.
- 65 H. Li, J. Shang, Z. H. Ai and L. Z. Zhang, *J. Am. Chem. Soc.*, 2015, **137**, 6393–6399.

