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# Tuning symmetry and magnetic blocking of an exchange-coupled lanthanide ion in isomeric, tetrametallic complexes: $[\text{LnCl}_6(\text{TiCp}_2)_3]^\dagger$

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The synthesis and magnetic properties of two pairs of isomeric, exchange-coupled complexes,  $[\text{LnCl}_6(\text{TiCp}_2)_3]$  (Ln = Gd, Tb), are reported. In each isomeric pair, the central lanthanide ion adopts either a pseudo-octahedral (O-Ln) or trigonal prismatic geometry (TP-Ln) yielding complexes with  $C_1$  or  $C_{3h}$  molecular symmetry, respectively. Ferromagnetic exchange coupling is observed in TP-Ln as indicated by the increases in  $\chi_m T$  below 30 K. For TP-Gd, a fit to the susceptibility reveals ferromagnetic coupling between the  $\text{Gd}^{3+}$  ion and the  $\text{Ti}^{3+}$  ions ( $J = 2.90(1) \text{ cm}^{-1}$ ). In contrast to O-Tb, which shows no single-molecule magnetic behavior, the TP-Tb complex presents slow magnetic relaxation with a 100s-blocking temperature of 2.3 K and remanent magnetization at zero field up to 3 K. The calculated electronic structures of both compounds imply that trigonal prismatic geometry of TP-Tb is critical to the observed magnetic behavior.

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## Introduction

f-Elements with high single-ion anisotropy have been identified as ideal building blocks for high-performance single ion magnets (SIMs) after the report of  $[\text{TbPc}_2]^-$  (Pc = phthalocyanine),<sup>1-4</sup> and have achieved remarkable blocking temperatures.<sup>5-7</sup> There are different strategies for reducing the rate of relaxation, and designing high-performance single-molecule magnets (SMMs), including using appropriate molecular geometry and the introduction of magnetic exchange coupling.<sup>8,9</sup> These exchange interactions span the range of exchange coupling with 3d metals and organic radical ligands, dipolar coupling, and direct exchange in lanthanide-lanthanide bonds.<sup>10-14</sup> Building exchange interactions into high-symmetry lanthanide SIMs, while maintaining the local symmetry of the lanthanide ion in the resultant exchange coupled SMM is a substantial synthetic challenge.<sup>9,14</sup> There are relatively few

examples and most cases require compromises on the local symmetry of the lanthanide ion.<sup>11,15,16</sup>

The appropriate molecular geometry can enhance the magnetic anisotropy and reduce the quantum tunneling of magnetization (QTM) process. For lanthanide cations sitting in certain local symmetry such as  $C_{5h}$  and  $C_n$  ( $n > 6$ ), the transverse crystal field parameters vanish, and therefore QTM process can be suppressed in SIMs.<sup>9,17</sup> Isomeric pairs are ideal for studying geometry and symmetry effects on SMM behaviors. However, few examples have been reported due to the significant synthetic challenge.<sup>18,19</sup> The magnetic exchange can create higher spin manifolds in molecular platforms and reduce the QTM relaxation.<sup>20</sup> Multinuclear lanthanide complexes featuring radical bridging ligands show magnetic blocking temperature (temperature where the relaxation time is 100 s) up to 20 K partially due to the magnetic interactions between spins.<sup>10,21</sup> Some heterometallic complexes with both lanthanide and transition metals are also found to be good SMMs.<sup>12,22</sup> Therefore, the combination of both appropriate ligand field and magnetic interaction can lead to high-performance f-element based SMMs with largely quenched QTM process and high single-ion anisotropy.<sup>7,11,16</sup>

Titanium(III) with a  $d^1$  electronic configuration can be used for the design of magnetic materials. The reports on the titanium-based molecular nanomagnets are limited which could be attributed to the instability of  $\text{Ti}^{3+}$  in ambient conditions. The recent studies on  $\text{Ti}^{3+}$  based qubits<sup>23,24</sup> and geometrically frustrated magnets<sup>25</sup> suggests that  $\text{Ti}^{3+}$  could serve as a good candidate for the design of novel quantum materials. However, to our knowledge, no SMMs with  $\text{Ti}^{3+}$  has been

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reported. In this case, we employ the small spin orbit coupling of the  $d^1$  electron in the  $a_1$  orbital (mixed  $s$  and  $d_{x^2-y^2}$  parentage) of the  $[\text{Cp}_2\text{TiCl}_2]^{1-}$  metalloligand to simplify the analysis of the exchange coupling between the titanium sites and the anisotropic lanthanide as a function of the site symmetry of the lanthanide ion.<sup>26</sup> This structure establishes a design and analytical framework to address molecular analogs of exchange coupling in electronically or geometrically frustrated 2D materials. To this end, we note that forcing a central anisotropic spin center to mutually satisfy ferromagnetic exchange with three metalloligands is a simple structural analog of the connectivity, and requisite dominant exchange pathways, in honeycomb materials such as  $\text{Na}_2\text{PrO}_3$ ,  $\text{Na}_2\text{IrO}_3$ , and  $\text{RuCl}_3$ ;<sup>27–29</sup> the latter two systems may have dominant Kitaev exchange.<sup>30</sup> This paradigm may eventually build to molecules in which all the exchange coupled ions present significant anisotropy and are closer analogs of the 2D materials.<sup>31</sup>

Herein we report the synthesis of two pairs of isomeric, lanthanide complexes,  $[\text{LnCl}_6(\text{TiCp}_2)_3]$ , ( $\text{Ln} = \text{Gd}, \text{Tb}$ ). The two isomeric pairs include a pseudo-octahedral geometry (**O-Ln**) and a trigonal prismatic geometry at the central lanthanide (**TP-Ln**). The **O-Tb** shows no SMM behaviors while the **TP-Tb** shows slow relaxation behavior with 100s blocking temperature of 2.3 K. The calculated electronic structures of both compounds suggest that trigonal prismatic geometry is critical to the observed magnetic behavior in **TP-Tb**.

## Results and discussion

The synthetic methodology was based on the preparation  $(\text{Cp}_2\text{Ti}^{\text{III}}\text{Cl})_2\text{MCl}_2$  ( $\text{Cp} = \text{C}_5\text{H}_5^-$ ,  $\text{M} = \text{Zn}, \text{Mn}$  and  $\text{Be}$ ) compounds which were prepared in Lewis acid/base reactions.<sup>32,33</sup> The **O-Tb** complex, was prepared in the reaction of anhydrous  $\text{TbCl}_3$  and  $(\text{Cp}_2\text{TiCl})_2$  ( $n_{\text{TbCl}_3} : n_{(\text{Cp}_2\text{TiCl})_2} = 2 : 3.05$ ) in toluene (82% yield). However, the reaction of  $(\text{Cp}_2\text{TiCl})_2$  with excess  $\text{TbCl}_3 \cdot x\text{THF}$  ( $n_{\text{TbCl}_3 \cdot x\text{THF}} : n_{(\text{Cp}_2\text{TiCl})_2} = 2.16 : 1$ ) results in the formation of both **O-Tb** and its isomeric analog, **TP-Tb**. Although the yield of

**TP-Tb** in the reaction is low ( $\sim 10\%$  yield), these two isomers crystallized independently with different crystal morphologies (Fig. 1 and S1,† needles of **TP-Tb** and parallelepipeds of **O-Tb**), which allowed the physical separation of the **TP-Tb** complex (see ESI†). The gadolinium analogs of these two terbium isomers, **TP-Gd** ( $\text{Gd}^{3+}$  in trigonal prismatic geometry) and **O-Gd** ( $\text{Gd}^{3+}$  in octahedral geometry), were also prepared. However, the operative conditions do not follow the Tb analogs. The reaction of  $\text{GdCl}_3 \cdot x\text{THF}$  with  $(\text{Cp}_2\text{TiCl})_2$  in toluene results only in **TP-Gd** (57% yield). The **O-Gd** can only be isolated from the reaction mixture (both **O-Gd** and **TP-Gd** were found) of the redox reaction between  $\text{Cp}_2\text{TiCl}_2$  and Gd powder. The yield of **O-Gd** is also low (less than  $\sim 5\%$  yield). All attempts to prepare  $[\text{DyCl}_6(\text{TiCp}_2)_3]$  with trigonal prismatic geometry were not successful despite that the Kramers ion  $\text{Dy}^{3+}$  could also result in interesting SMM behavior. Synthetic attempts on different lanthanides suggest that the formation of **TP-Ln** or **O-Ln** phases largely depends on the size of lanthanide cations. For  $\text{Gd}^{3+}$ , which has a larger ionic radius, the **TP-Gd** is the stable phase which is of high yield. For the smaller  $\text{Tb}^{3+}$  and  $\text{Dy}^{3+}$  cations, the **TP-Ln** phases are of extremely low yield and the **O-Tb** is the dominant product in reactions.

The structures of these four compounds were characterized by single-crystal X-ray diffraction (Fig. 1, S3 and S4†) and can be described by the chelation of three  $[\text{Cp}_2\text{Ti}^{\text{III}}\text{Cl}_2]^{1-}$  ligands to one  $\text{Ln}^{3+}$  cation. Both **TP-Tb** and **TP-Gd** crystallized into the  $P6_3/m$  space group, and the symmetry of both molecules can be described by the  $C_{3h}$  point group. SHAPE analysis indicates a slightly distorted trigonal prismatic geometry for the  $[\text{LnCl}_6]^{3-}$  polyhedron (Table S9†) in **TP-Ln**.<sup>34–36</sup> The **O-Tb** and **O-Gd** complexes crystallized into the  $P2_1/n$  space group. The  $[\text{LnCl}_6]^{3-}$  polyhedra in **O-Tb** and **O-Gd** have distorted octahedral geometry, and the symmetry of both molecules is  $C_1$ .

Due to the limited yield of **O-Gd**, only the single crystal X-ray diffraction (SCXRD), infrared (IR) and Raman spectroscopy data were collected on **O-Gd**. For **TP-Tb**, **O-Tb**, and **TP-Gd**, the purity of these samples was confirmed by multiple techniques. The elemental analysis (EA) of these three samples matched well with the theoretical value (see ESI† for details). However, the EA results can not differentiate the phase purity of isomers. The IR spectra of these samples are very similar as shown in Fig. S7a.† Notable differences between **TP-Ln** and **O-Ln** can be observed in Raman spectra (Fig. S7b†) which uniquely determines the phase purity sample on single crystal level. To determine bulk phase purities, power X-ray diffraction (PXRD) was performed. The PXRD patterns of **TP-Tb**, **O-Tb** and **TP-Gd** matched well with the simulated patterns from the SCXRD results, and no additional peaks can be observed for all patterns which indicates the bulk phase purities of these three samples (Fig. S6†).

dc Magnetic susceptibility data were collected for **TP-Tb**, **O-Tb**, and **TP-Gd** from 2 to 300 K under 0.1 T (Fig. 2). The  $\chi_{\text{M}}T$  ( $\chi_{\text{M}}$  is molar magnetic susceptibility) values at 300 K for **TP-Tb** and **O-Tb** are 13.35 and 12.06  $\text{emu K mol}^{-1}$  respectively. The theoretical value for one magnetically isolated  $\text{Tb}^{3+}$  and three  $S = 1/2$   $\text{Ti}^{3+}$  ions is 12.93  $\text{emu K mol}^{-1}$ . The  $\chi_{\text{M}}T$  value of **O-Tb** is slightly lower than the theoretical value. The  $\chi_{\text{M}}T$  value at 300 K for **TP-Gd** is 9.28  $\text{emu K mol}^{-1}$  which is close to the theoretical value of

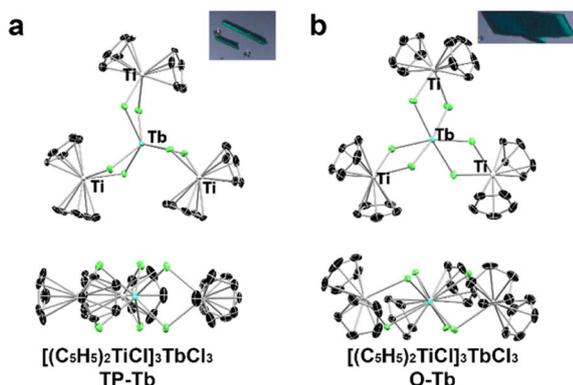


Fig. 1 (a) Molecular structure of **TP-Tb** with thermal ellipsoids at 50% probability. (Inset) Image of **TP-Tb** single crystals. (b) Molecular structure of **O-Tb** with thermal ellipsoids at 50% probability. (Inset) Image of **O-Tb** single crystals. Black, blue, green, and grey represent carbon, terbium, chlorine, and titanium, respectively.



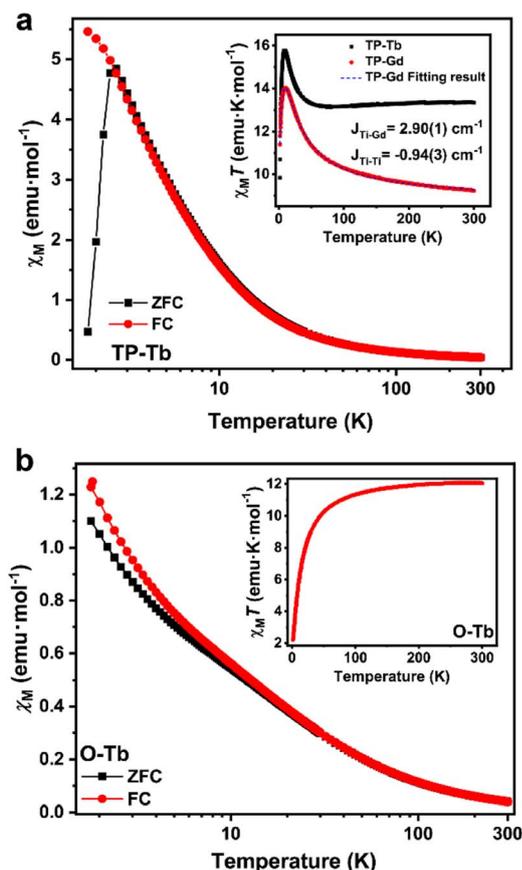


Fig. 2 (a) Temperature dependence of  $\chi_M$  at 0.1 T FC and ZFC conditions for TP-Tb. (Inset) Temperature dependence of  $\chi_M T$  at 0.1 T FC condition for TP-Tb and TP-Gd. The blue dashed line is the fitting result to the TP-Gd  $\chi_M T$  curve. (b) Temperature dependence of  $\chi_M$  at 0.1 T FC and ZFC condition for O-Tb. (Inset) Temperature dependence of  $\chi_M T$  at 0.1 T FC condition for O-Tb.

9.00 emu K mol<sup>-1</sup> for one magnetically isolated Gd<sup>3+</sup> ion and three Ti<sup>3+</sup> ions. For both TP-Tb and TP-Gd, with lower temperature,  $\chi_M T$  value starts to rise at 30 K, maximizing around 14 K. The maximum  $\chi_M T$  value for TP-Gd is 14.0 emu K mol<sup>-1</sup> which is slightly lower than the theoretical value of 15 emu K mol<sup>-1</sup> for an  $S = 5$  system which establishes the ferromagnetic interactions between Ti<sup>3+</sup> and Gd<sup>3+</sup>. This ferromagnetic interaction is indicated by the coupling constant between Ti<sup>3+</sup> and Gd<sup>3+</sup>,  $J_{\text{Ti-Gd}} = 2.90(1)$  cm<sup>-1</sup>, which was derived by fitting the temperature dependence of  $\chi_M T$  and the magnetic field dependence of magnetization of TP-Gd results.<sup>37</sup> Antiferromagnetic interactions between Ti<sup>3+</sup> sites are confirmed by the  $J_{\text{Ti-Ti}} = -0.94(3)$  cm<sup>-1</sup>. Considering the similar temperature dependence behavior observed in TP-Tb compared to TP-Gd, the magnetic interaction between Ti<sup>3+</sup> ions and the Tb<sup>3+</sup> ion are assumed to also be ferromagnetic with a similar ordering of the exchange coupling strength.<sup>38</sup>

For O-Tb, the  $\chi_M T$  value decreases with lower temperature from 300 K to 2 K. The lack of low-temperature maximum  $\chi_M T$  compared to the TP-Tb is indicative of different magnetic interactions between Tb<sup>3+</sup> and Ti<sup>3+</sup>. The temperature dependence of

$\chi_M$  for TP-Tb and O-Tb shows a dramatic difference. In O-Tb, the field-cooling (FC) and zero-field-cooling (ZFC) curves both increase monotonically with lower temperature from 300 K to 2 K. In TP-Tb, the FC curve increases monotonically with lower temperature while the ZFC curve peaks at 2.6 K. The divergence between FC and ZFC plot and the peak around 2.6 K in TP-Tb are indicative of the blocking of the magnetization and indicate that TP-Tb should present slow magnetic relaxation behavior.

The ac magnetic susceptibility measurement was conducted on both O-Tb and TP-Tb to study their dynamic magnetic properties. The data for O-Tb show negligible out-of-phase magnetic susceptibility ( $\chi_M''$ ) from 2 K to 9 K, which confirms the lack of slow magnetic relaxation behavior in O-Tb (Fig. S16<sup>†</sup>). For TP-Tb, one single peak can be observed in the frequency range from 1 to 750 Hz between 4 K to 30 K in the temperature dependence of  $\chi_M''$  plot (Fig. S10<sup>†</sup>). The peak position is temperature-dependent which suggests the lack of QTM process from 4 K to 30 K. Magnetic relaxation times,  $\tau$ , were extracted from fits of the ac susceptibility data to a generalized Debye model (Fig. 3a, see ESI<sup>†</sup>). The QTM process is temperature-independent and the temperature-dependence of  $\tau$  value should be flattened if the relaxation is governed by a fast QTM process. The  $\tau$  value increases with lower temperature down to 2 K which confirms the lack of fast QTM relaxation process in the probed range. Under zero magnetic field, without a fast QTM process, the relaxation behaviors are normally dominated by Orbach and Raman processes. However, the attempts to fit the  $\tau$  value from 6 K to 30 K with the inclusion of Raman processes lead to less reasonable fitting results (see ESI<sup>†</sup> for details). Fitting the  $\tau$  from 4 K to 30 K to the function  $\tau^{-1} = \tau_{0,1}^{-1} \exp(-U_{\text{eff},1}/k_B T) + \tau_{0,2}^{-1} \exp(-U_{\text{eff},2}/k_B T)$  results in  $U_{\text{eff},1} = 23.54(9)$  cm<sup>-1</sup>,  $\tau_{0,1} = 0.00152(2)$  s,  $U_{\text{eff},2} = 395(18)$  cm<sup>-1</sup>,  $\tau_{0,2} = 6.4(59) \times 10^{-12}$  s. The  $\tau$  value from 2 K to 4 K was derived by dc relaxation measurements. The  $\tau = 175(16)$  s at 2.2 K and  $\tau = 72(18)$  s at 2.4 K suggests the blocking temperature,  $T_B$ , (temperature where relaxation time  $\tau$  value is larger than 100 s), is around 2.3 K.

It is notable that the pre-factor  $\tau_0 = 0.00152(2)$  s for the Orbach process in TP-Tb is several orders higher than in most of the reported SMMs (Table S15<sup>†</sup>). Great progress has been made recently on improving the  $U_{\text{eff}}$  value, which is important for the slower Orbach relaxation process ( $\tau^{-1} = \tau_0^{-1} \exp(-U_{\text{eff}}/k_B T)$ ).<sup>5,6</sup> However, increasing the value of  $\tau_0$ , which is related to the vibrational modes of molecules, remains challenging.<sup>39,40</sup> Recent theoretical works suggest that further improvements to monometallic single-molecule magnets requires moving vibrational modes off-resonance with the electronic transitions.<sup>41-43</sup> The variable field magnetization measurements on TP-Tb at a sweep rate of 40 Oe s<sup>-1</sup> when  $|\mu_0 H| < 2$  T and 150 Oe s<sup>-1</sup> when  $|\mu_0 H| > 2$  T reveal magnetic hysteresis loop with remnant magnetization at zero field up to 3 K (Fig. 3b), indicating that TP-Tb is a hard-magnet below 3 K. The sharp drop of coercive field at 2.5 K compared to 2 K could be attributed to the Orbach relaxation procedure with  $U_{\text{eff},1} = 23.54(9)$  cm<sup>-1</sup> since the  $\tau_{0,1}^{-1} \exp(-U_{\text{eff},1}/k_B T)$  equals to 1163 s at 2.5 K and equals to  $3.4 \times 10^4$  s at 2 K.

The in-field ac susceptibility measurements on TP-Tb were also conducted. As shown in Fig. S13-S15,<sup>†</sup> the application of



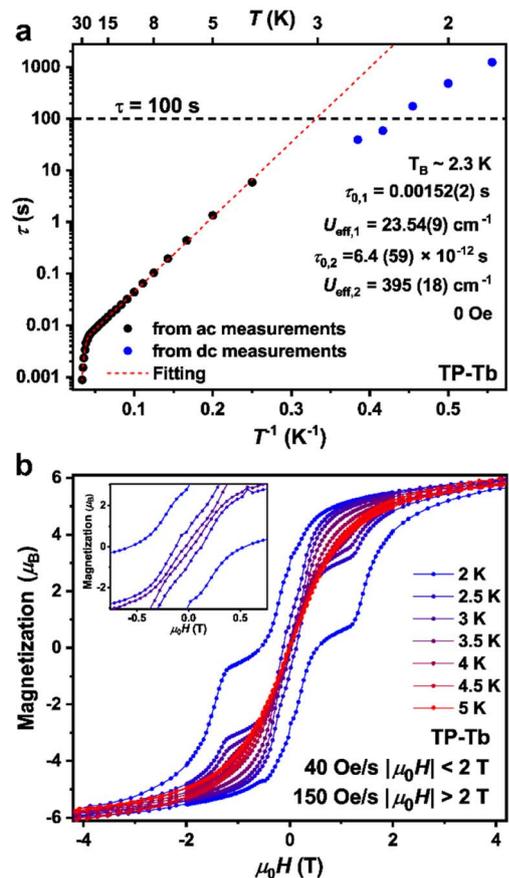


Fig. 3 (a) Inverse temperature dependence of  $\tau$ . The black dots are derived from fitting the  $\chi_M'$  dependence of  $\chi_M''$  plot (Cole–Cole plot) to the generalized Debye model, while the blue dots are derived from fitting the dc relaxation data to the stretched exponential functions. The red dashed line corresponds to the fits of  $\tau$  from 8 K to 30 K to the function  $\tau^{-1} = \tau_{0,1}^{-1} \exp(-U_{\text{eff},1}/k_B T) + \tau_{0,2}^{-1} \exp(-U_{\text{eff},2}/k_B T)$ . The black dashed line corresponds to the line  $\tau = 100$  s. (b) Variable field magnetization data for TP-Tb collected from 2 K to 5 K at a sweep rate of 40 Oe s $^{-1}$  when  $|\mu_0 H| < 2$  T and 150 Oe s $^{-1}$  when  $|\mu_0 H| > 2$  T.

magnetic field decreases the relaxation time especially with the application of 2500 Oe dc magnetic field. Only QTM and direct processes are magnetic field dependent.<sup>8,9</sup> The QTM can be suppressed by higher field and yield increased relaxation times, while the direct process can be amplified by applied field leading to decreased relaxation times. Therefore, the decrease of relaxation time with higher magnetic field is consistent with our conclusion for the lack of a dominant QTM process in TP-Tb in the probed range and is indicative of the existence of a direct process under an applied magnetic field for TP-Tb.

To gain insight into the origin of the magnetic properties of the O-Tb and TP-Tb complexes, we performed XMS-CASPT2 electronic structure calculations on the reduced-size model complex, [TbCl<sub>6</sub>]<sup>3-</sup>, using the OpenMolcas software<sup>44</sup> (see ESI<sup>†</sup>). The energy diagrams of the electronic states arising from the crystal field splitting of the ground  $^7F_6$  level of Tb<sup>3+</sup> ion and the transition magnetic dipole moments between these states are shown in Fig. 4. The electronic states of TP-Tb are nearly doubly-degenerate, the energy of the first excited state (260 cm $^{-1}$ ) and

the anisotropy barrier (446 cm $^{-1}$ ) are large compared to those of O-Tb (41 cm $^{-1}$  and 363 cm $^{-1}$ , respectively). These differences in electronic structure reflect the  $C_{3h}$  symmetry of the TP-Tb complex and the strongly distorted geometry of O-Tb. Meanwhile, the transition dipoles indicate that in TP-Tb the magnetic relaxation proceeds through few high-energy excited states, while in O-Tb many low-energy states are involved. The larger transition dipoles in O-Tb complex arises from the significant mixing of the  $M_J$  components in each electronic state (Table S16<sup>†</sup>). In contrast, the TP-Tb electronic states are nearly pure superpositions of  $\pm M_J$  components (Table S17<sup>†</sup>). These computational results are in agreement with the experimental finding of slow magnetic relaxation in the TP-Tb complex and the absence of SMM behaviors in O-Tb.

The calculated energy of first excited doublet (260 cm $^{-1}$ ) is much larger than the  $U_{\text{eff},1}$  (23.5 cm $^{-1}$ ) but closer to  $U_{\text{eff},2}$  (395 cm $^{-1}$ ) in TP-Tb. Considering that the  $J_{\text{Tl-Gd}} = 2.90(1)$  cm $^{-1}$  in TP-Gd, it is suggestive that the dominant Orbach process is dictated by the strength of exchange coupling despite the strong anisotropy of Tb<sup>3+</sup> in TP-Tb. Similar phenomenon has been studied in several radical-bridged dylanthanide complexes.<sup>38</sup> Although magnetic exchange leads to an Orbach process with a small barrier ( $U_{\text{eff},1} = 23.5$  cm $^{-1}$ ) in TP-Tb, the exchange coupling greatly suppresses the zero-field relaxation behavior (mainly the QTM process) since the monolanthanide-[1] metallocenophane with similar  $D_{3h}$  Ln<sup>3+</sup> local structure, but no magnetic exchange, show soft magnetism and butterfly-shape hysteresis loops.<sup>45,46</sup>

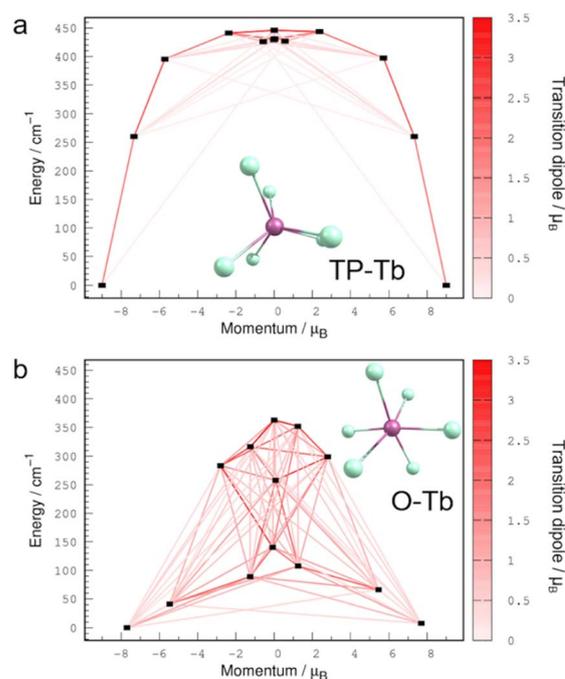


Fig. 4 The energy diagrams of the electronic states arising from the crystal field splitting of the ground  $^7F_6$  level of Tb<sup>3+</sup> ion and the transition magnetic dipole moments between these states calculated for the models of TP-Tb (a) and O-Tb (b) complexes.



## Conclusions

In conclusion, the magnetic property studies on these newly synthesized species show that the **TP-Tb** complex exhibits SMM behaviors with the blocking temperature of 2.3 K and remanent magnetization at zero field up to 3 K, while the **O-Tb** complex shows fast magnetic relaxation behavior in the probed range. The observed properties are attributable to the combination of high local symmetry of the lanthanide ion and the magnetic exchange between  $Ti^{3+}$  and  $Tb^{3+}$  in **TP-Tb**. This work builds on a nascent strategy to improve SMMs by combining the magnetic exchange interactions and high local symmetry in f-element based molecules. Additionally, these complexes establish a step-wise path to examining the role of geometric and electronic magnetic frustration in the dynamic properties of molecular nanomagnets.

## Data availability

The data supporting the findings of this study are available within the article and in the ESI.†

## Author contributions

N. J. and H. S. L. conceived the project. N. J. carried out the synthesis; D. N., and S. V. performed the theoretical analyses; J. X., and N. J. performed and analyzed the magnetic property measurements; N. J., H. S. L., D. N., and S. V. wrote the paper. All authors revised the manuscript and discussed the results.

## Conflicts of interest

There are no conflicts to declare.

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## Notes and references

- 1 N. Ishikawa, M. Sugita, T. Ishikawa, S.-y. Koshihara and Y. Kaizu, *J. Am. Chem. Soc.*, 2003, **125**, 8694–8695.
- 2 J. D. Rinehart and J. R. Long, *Chem. Sci.*, 2011, **2**, 2078–2085.
- 3 C. A. Goodwin, *Dalton Trans.*, 2020, **49**, 14320–14337.

- 4 D. N. Woodruff, R. E. Winpenny and R. A. Layfield, *Chem. Rev.*, 2013, **113**, 5110–5148.
- 5 C. A. Goodwin, F. Ortu, D. Reta, N. F. Chilton and D. P. Mills, *Nature*, 2017, **548**, 439–442.
- 6 F.-S. Guo, B. M. Day, Y.-C. Chen, M.-L. Tong, A. Mansikkamäki and R. A. Layfield, *Science*, 2018, **362**, 1400–1403.
- 7 C. A. Gould, K. R. McClain, D. Reta, J. G. Kragoskow, D. A. Marchiori, E. Lachman, E.-S. Choi, J. G. Analytis, R. D. Britt and N. F. Chilton, *Science*, 2022, **375**, 198–202.
- 8 S. T. Liddle and J. van Slageren, *Chem. Soc. Rev.*, 2015, **44**, 6655–6669.
- 9 J.-L. Liu, Y.-C. Chen and M.-L. Tong, *Chem. Soc. Rev.*, 2018, **47**, 2431–2453.
- 10 J. D. Rinehart, M. Fang, W. J. Evans and J. R. Long, *Nat. Chem.*, 2011, **3**, 538–542.
- 11 S.-G. Wu, Z.-Y. Ruan, G.-Z. Huang, J.-Y. Zheng, V. Vieru, G. Taran, J. Wang, Y.-C. Chen, J.-L. Liu and L. F. Chibotaru, *Chem*, 2021, **7**, 982–992.
- 12 L. E. Darago, M. D. Boshart, B. D. Nguyen, E. Perlt, J. W. Ziller, W. W. Lukens, F. Furché, W. J. Evans and J. R. Long, *J. Am. Chem. Soc.*, 2021, **143**, 8465–8475.
- 13 Y.-N. Guo, G.-F. Xu, W. Wernsdorfer, L. Ungur, Y. Guo, J. Tang, H.-J. Zhang, L. F. Chibotaru and A. K. Powell, *J. Am. Chem. Soc.*, 2011, **133**, 11948–11951.
- 14 Y.-C. Chen and M.-L. Tong, *Chem. Sci.*, 2022, **13**, 8716–8726.
- 15 J. D. Rinehart, M. Fang, W. J. Evans and J. R. Long, *J. Am. Chem. Soc.*, 2011, **133**, 14236–14239.
- 16 P. Zhang, F. Benner, N. F. Chilton and S. Demir, *Chem*, 2022, **8**, 717–730.
- 17 Y.-C. Chen, J.-L. Liu, L. Ungur, J. Liu, Q.-W. Li, L.-F. Wang, Z.-P. Ni, L. F. Chibotaru, X.-M. Chen and M.-L. Tong, *J. Am. Chem. Soc.*, 2016, **138**, 2829–2837.
- 18 C.-H. Chen, D. S. Krylov, S. M. Avdoshenko, F. Liu, L. Spree, R. Yadav, A. Alvertis, L. Hozoi, K. Nenkov and A. Kostanyan, *Chem. Sci.*, 2017, **8**, 6451–6465.
- 19 J.-L. Liu, Y.-C. Chen, Y.-Z. Zheng, W.-Q. Lin, L. Ungur, W. Wernsdorfer, L. F. Chibotaru and M.-L. Tong, *Chem. Sci.*, 2013, **4**, 3310–3316.
- 20 S. Demir, I.-R. Jeon, J. R. Long and T. D. Harris, *Coord. Chem. Rev.*, 2015, **289**, 149–176.
- 21 S. Demir, M. I. Gonzalez, L. E. Darago, W. J. Evans and J. R. Long, *Nat. Commun.*, 2017, **8**, 1–9.
- 22 K. Liu, W. Shi and P. Cheng, *Coord. Chem. Rev.*, 2015, **289**, 74–122.
- 23 S. Von Kugelgen, M. D. Krzyaniak, M. Gu, D. Puggioni, J. M. Rondinelli, M. R. Wasielewski and D. E. Freedman, *J. Am. Chem. Soc.*, 2021, **143**, 8069–8077.
- 24 L. C. de Camargo, M. Briganti, F. S. Santana, D. Stingen, R. R. Ribeiro, G. G. Nunes, J. F. Soares, E. Salvadori, M. Chiesa and S. Benci, *Angew. Chem.*, 2021, **133**, 2620–2625.
- 25 N. Jiang, A. Ramanathan, J. Bacsá and H. S. La Pierre, *Nat. Chem.*, 2020, **12**, 691–696.
- 26 T. A. Albright, J. K. Burdett and M.-H. Whangbo, *Orbital Interactions In Chemistry*, John Wiley & Sons, 2013.



- 27 M. J. Daum, A. Ramanathan, A. I. Kolesnikov, S. Calder, M. Mourigal and H. S. La Pierre, *Phys. Rev. B: Condens. Matter*, 2021, **103**, L121109.
- 28 V. M. Katukuri, S. Nishimoto, V. Yushankhai, A. Stoyanova, H. Kandpal, S. Choi, R. Coldea, I. Rousochatzakis, L. Hozoi and J. Van Den Brink, *New J. Phys.*, 2014, **16**, 013056.
- 29 R. Yadav, N. A. Bogdanov, V. M. Katukuri, S. Nishimoto, J. Van Den Brink and L. Hozoi, *Sci. Rep.*, 2016, **6**, 1–16.
- 30 A. Kitaev, *Ann. Phys.*, 2006, **321**, 2–111.
- 31 F. P. Liu, H. Gornitzka, D. Stallze and H. W. Roesky, *Angew. Chem.*, 1993, **105**, 447–448.
- 32 R. Coutts, P. Wailes and R. Martin, *J. Organomet. Chem.*, 1973, **47**, 375–382.
- 33 D. G. Sekutowski and G. D. Stucky, *Inorg. Chem.*, 1975, **14**, 2192–2199.
- 34 M. Pinsky and D. Avnir, *Inorg. Chem.*, 1998, **37**, 5575–5582.
- 35 D. Casanova, J. Cirera, M. Llunell, P. Alemany, D. Avnir and S. Alvarez, *J. Am. Chem. Soc.*, 2004, **126**, 1755–1763.
- 36 J. Cirera, E. Ruiz and S. Alvarez, *Chem.–Eur. J.*, 2006, **12**, 3162–3167.
- 37 N. F. Chilton, R. P. Anderson, L. D. Turner, A. Soncini and K. S. Murray, *J. Comput. Chem.*, 2013, **34**, 1164–1175.
- 38 C. A. Gould, E. Mu, V. Vieru, L. E. Darago, K. Chakarawet, M. I. Gonzalez, S. Demir and J. R. Long, *J. Am. Chem. Soc.*, 2020, **142**, 21197–21209.
- 39 A. Lunghi, F. Totti, R. Sessoli and S. Sanvito, *Nat. Commun.*, 2017, **8**, 1–7.
- 40 A. Castro-Alvarez, Y. Gil, L. Llanos and D. Aravena, *Inorg. Chem. Front.*, 2020, **7**, 2478–2486.
- 41 D. Reta, J. G. Kragoskow and N. F. Chilton, *J. Am. Chem. Soc.*, 2021, **143**, 5943–5950.
- 42 A. Ullah, J. Cerdá, J. J. Baldoví, S. A. Varganov, J. Aragón and A. Gaita-Arino, *J. Phys. Chem. Lett.*, 2019, **10**, 7678–7683.
- 43 V. D. Dergachev, D. D. Nakritskaia and S. A. Varganov, *J. Phys. Chem. Lett.*, 2022, **13**, 6749–6754.
- 44 I. F. Galván, M. Vacher, A. Alavi, C. Angeli, F. Aquilante, J. Autschbach, J. J. Bao, S. I. Bokarev, N. A. Bogdanov and R. K. Carlson, *J. Chem. Theory Comput.*, 2019, **15**, 5925–5964.
- 45 T. P. Latendresse, N. S. Bhuvanesh and M. Nippe, *J. Am. Chem. Soc.*, 2017, **139**, 8058–8061.
- 46 T. P. Latendresse, V. Vieru, B. O. Wilkins, N. S. Bhuvanesh, L. F. Chibotaru and M. Nippe, *Angew. Chem.*, 2018, **130**, 8296–8301.

