


 Cite this: *RSC Adv.*, 2023, **13**, 26144

## Controlled synthesis of $\text{Mo}_2\text{C}$ micron flowers *via* vapor–liquid–solid method as enhanced electrocatalyst for hydrogen evolution reaction

 Yuwei Wang,<sup>a</sup> Jian He,<sup>b</sup> Yipeng Zang,<sup>c</sup> Changbao Zhao,<sup>c</sup> Miaomiao Di<sup>b</sup> and Bin Wang  <sup>\*bc</sup>

$\text{Mo}_2\text{C}$  demonstrates excellent performance in catalysis, and it has been found to possess excellent hydrogen evolution reaction (HER) catalytic activity and highly efficient nitrogen fixation. The catalytic activity of  $\text{Mo}_2\text{C}$  is greatly influenced and restricted by the preparation method. Sintering and carbon deposition, which affect the catalytic activity of  $\text{Mo}_2\text{C}$ , are inevitable in the traditional vapor–solid–solid (VSS) process. In this study, we report the controllable synthesis of  $\alpha$ - $\text{Mo}_2\text{C}$  micron flowers by adjusting the growth temperature *via* a vapor–liquid–solid (VLS) process. The density of the  $\text{Mo}_2\text{C}$  micron flowers is closely related to the concentration of  $\text{Na}_2\text{MoO}_4$  aqueous solution. The as-grown  $\text{Mo}_2\text{C}$  micron flowers dispersed with Pt are validated to be an enhanced collaborative electrocatalyst for HER against Pt/VSS- $\text{Mo}_2\text{C}$ .

Received 18th July 2023

Accepted 18th August 2023

DOI: 10.1039/d3ra04813f

[rsc.li/rsc-advances](http://rsc.li/rsc-advances)

## Introduction

In recent years, transition metal carbides (TMCs) have been intensively researched for their specific chemical properties.<sup>1–3</sup> TMCs are considered to be similar to precious metals in the aspect of electrochemistry and catalysis.  $\text{Mo}_2\text{C}$  belonging to the TMC family is called quasi platinum catalyst, and it plays an important role in highly efficient nitrogen fixation<sup>4</sup> and hydrogen evolution reaction (HER).<sup>5–9</sup> Especially, as some metal atoms are dispersed on the surface of  $\text{Mo}_2\text{C}$  crystals for collaborative catalysis, they exhibit excellent selectivity and superior activity for many catalytic reactions.<sup>10–13</sup> However, many collaborative catalysis suffer from low mass-specific activity owing to the low metal loading.<sup>14</sup> In order to optimize metal loading, the support  $\text{Mo}_2\text{C}$  crystals should have a high specific surface area which can provide abundant surface sites to enhance the collaborative catalysis.

As a catalytic material with excellent performance, the catalytic activity of  $\text{Mo}_2\text{C}$  is greatly influenced and restricted by its preparation method.<sup>15–18</sup> In earlier studies, the sintering of the as-grown  $\text{Mo}_2\text{C}$  crystals was inevitable,<sup>19,20</sup> influencing the structure and morphology of  $\text{Mo}_2\text{C}$  crystals, which results in the reduction of the specific surface area and catalytic activity. Therefore, it is important to improve the preparation methods

to reduce the sintering and thus increase the specific surface area of the  $\text{Mo}_2\text{C}$  crystals.

Herein, we report the synthesis of  $\alpha$ - $\text{Mo}_2\text{C}$  crystals *via* an atmospheric pressure vapor–liquid–solid (VLS) method with  $\text{Na}_2\text{MoO}_4$  as the Mo precursor. The morphology of the  $\text{Mo}_2\text{C}$  crystals could be controlled by adjusting the growth temperature.  $\text{Mo}_2\text{C}$  micron flowers were obtained when the growth temperature was 780 °C. Compared with the vapor–solid–solid (VSS) mode, VLS mode has the advantages of good wettability and superior mobility, which can promote the lateral migration of Mo precursors and prevent the reactive materials from accumulating.<sup>11–25</sup> Thus, the as-grown  $\text{Mo}_2\text{C}$  crystals can form sheet morphology at an appropriate temperature comparing with the block morphology formation at higher temperatures or *via* the VSS mode. The advantage of VLS over VSS mode can be further demonstrated by comparing the HER catalytic activity of the as-grown  $\text{Mo}_2\text{C}$  dispersed with Pt. Pt/VLS- $\text{Mo}_2\text{C}$  has a lower overpotential than Pt/VSS- $\text{Mo}_2\text{C}$  at a current density of 10 mA cm<sup>−2</sup>.  $\text{Mo}_2\text{C}$  crystals grown using the VLS method is of great significance to improve their catalytic activity and expand their application fields.

## Results and discussion

The CVD growth process of  $\text{Mo}_2\text{C}$  on Au substrate is illustrated in Fig. 1a. The upper panel shows a typical VSS mode for the growth where  $(\text{NH}_4)_6\text{Mo}_7\text{O}_{24}$  aqueous solution is used as the Mo precursor. As the growth temperature reaches 780 °C,  $(\text{NH}_4)_6\text{Mo}_7\text{O}_{24}$  decomposed to form the solid state of  $\text{MoO}_3$  particles, which were then carbonized to produce  $\text{Mo}_2\text{C}$  when  $\text{C}_2\text{H}_4$  was introduced into the reaction chamber. Fig. 1b shows the

<sup>a</sup>College of Physical Science and Technology, Bohai University, Jinzhou, 121013, China

<sup>b</sup>College of Chemistry and Materials Engineering, Bohai University, Jinzhou, 121013, China. E-mail: wangbinhx@163.com

<sup>c</sup>State Key Laboratory of Catalysis, Dalian Institute of Chemical Physics, Chinese Academy of Sciences, Dalian, 116023, China

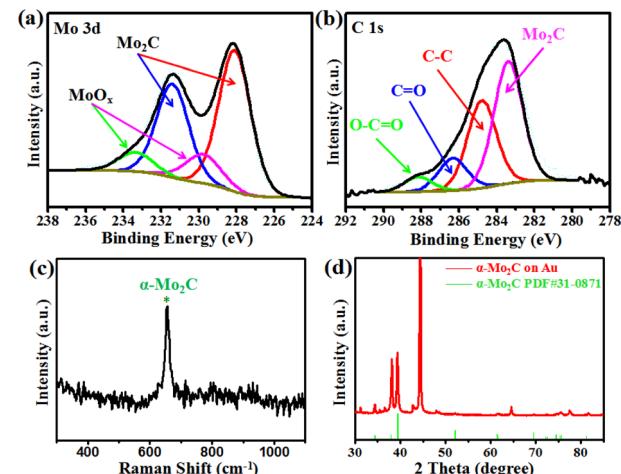
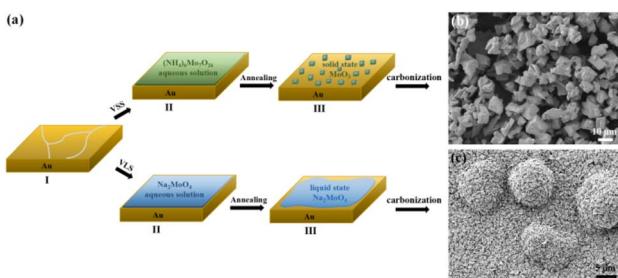


SEM image of the as-grown  $\text{Mo}_2\text{C}$  with 150 mg per mL  $(\text{NH}_4)_6\text{Mo}_7\text{O}_{24}$  as the Mo precursor. The  $\text{Mo}_2\text{C}$  demonstrated block morphology with size inconsistency.

$\text{Mo}_2\text{C}$  micron flowers with high specific surface area were synthesized *via* the VLS mode, and the schemes are shown in the bottom panel of Fig. 1a. 150 mg per mL  $\text{Na}_2\text{MoO}_4$  aqueous solution replaces  $(\text{NH}_4)_6\text{Mo}_7\text{O}_{24}$  aqueous solution as the Mo precursor. It is worth noting that the melting point of  $\text{Na}_2\text{MoO}_4$  is 687 °C, it melts into liquid state and forms a liquid–solid interface with Au substrate at the growth temperature (780 °C). Importantly, liquid has the advantage of a lower migration barrier, which is more beneficial to the unrestricted diffusion and homogeneous distribution of the precursors on the Au substrate. Thus, uniform  $\text{Mo}_2\text{C}$  micron sheets can be synthesized *via* the VLS mode. Moreover, the liquid–solid interface is conducive to the lateral growth of  $\text{Mo}_2\text{C}$  micron sheets. As the size and density increases, the  $\text{Mo}_2\text{C}$  micron sheets gradually form the  $\text{Mo}_2\text{C}$  micron flower morphology, as shown in Fig. 1c.

X-ray photoelectron spectroscopy (XPS) was conducted to evaluate the chemical composition and valence state of the  $\text{Mo}_2\text{C}$  crystals. Fig. 2a shows the binding energies of Mo 3d peaks at 231.4 eV and 228.1 eV, which are attributed to the Mo 3d<sub>3/2</sub> and Mo 3d<sub>5/2</sub>, respectively.<sup>26–30</sup> In addition, two weak peaks were observed at 233.4 and 229.8 eV, representing the intermediate oxidation states of Mo ( $\text{MoO}_x$ ).<sup>28,29</sup> The  $\text{MoO}_x$  may have resulted either from the exposure of  $\text{Mo}_2\text{C}$  to air or from the oxidization of  $\text{Mo}_2\text{C}$  during the XPS measurement process. Fig. 2b shows the C 1s XPS spectrum, whereby the peak located at the lower binding energy of 283.3 eV was assigned to C–Mo,<sup>26,27,30</sup> and those peaks at higher binding energies of 284.8, 286.3, and 288.1 eV can be ascribed to the carbons in the non-oxygenated C–C, C=O, and O=C=O, respectively.<sup>28,29</sup> The XPS signals confirmed the identity of the  $\text{Mo}_2\text{C}$  crystals, as expected. Raman spectroscopy and XRD were conducted to evaluate the structure of the  $\text{Mo}_2\text{C}$  crystals (Fig. 2c and d). Raman spectrum showed a well-defined characteristic peak at 652 cm<sup>−1</sup>, corresponding to the  $\text{A}_\text{g}$  mode of  $\alpha\text{-Mo}_2\text{C}$  crystal.<sup>31,32</sup> The diffraction peaks of  $\text{Mo}_2\text{C}$  in the X-ray diffraction (XRD) spectra were consistent with the standard XRD pattern of  $\text{Mo}_2\text{C}$  (PDF#31-0871), demonstrating that the as-grown  $\text{Mo}_2\text{C}$  crystals were  $\alpha\text{-Mo}_2\text{C}$ .

The morphology and density of  $\text{Mo}_2\text{C}$  crystals can be tuned remarkably by changing the growth temperature and the concentration of  $\text{Na}_2\text{MoO}_4$  aqueous solution. Fig. 3a and



b present the SEM images of the  $\text{Mo}_2\text{C}$  micron sheets grown with 30 and 75 mg per mL  $\text{Na}_2\text{MoO}_4$  aqueous solutions at 780 °C, respectively. When the  $\text{Na}_2\text{MoO}_4$  aqueous concentration was 30 mg mL<sup>−1</sup>, it provided a low concentration of Mo species, resulting in few nucleation sites, and thus, only a low quantity of  $\text{Mo}_2\text{C}$  micron sheets appeared, as shown in Fig. 3a. By increasing the  $\text{Na}_2\text{MoO}_4$  aqueous concentration to 75 mg mL<sup>−1</sup>, the shape of the  $\text{Mo}_2\text{C}$  micron sheets became more evident, whereby some micron sheets have begun to form flower-like shapes. The inset in Fig. 3b is the SEM image of an individual  $\text{Mo}_2\text{C}$  micron flower. Energy dispersive X-ray spectroscopy (EDS) mapping were recorded for the spatial distribution of the Mo and C elements (Fig. 3c and d), and both of them were found to be distributed uniformly in the micron flowers with sharp edges, exhibiting the uniformity of the  $\text{Mo}_2\text{C}$  crystals. Subsequently, the influence of growth temperature was investigated, and the 75 mg per mL  $\text{Na}_2\text{MoO}_4$  aqueous solution was used as

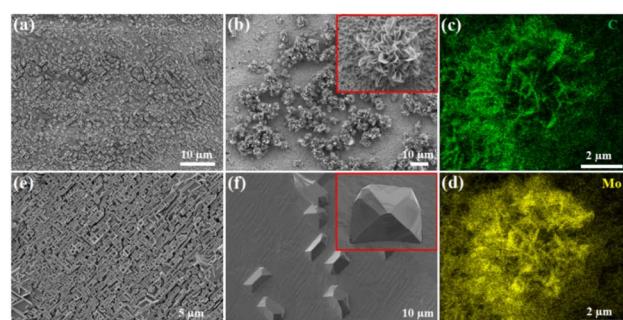


Fig. 3 SEM images of the  $\text{Mo}_2\text{C}$  crystals grown with different concentrations of  $\text{Na}_2\text{MoO}_4$  aqueous solution: (a) 30 mg mL<sup>−1</sup> and (b) 75 mg mL<sup>−1</sup>. (c) and (d) EDS elemental mapping of Mo and C of the  $\text{Mo}_2\text{C}$  micron flower, respectively. SEM images of the  $\text{Mo}_2\text{C}$  crystals grown with different temperatures: (e) 850 °C and (f) 900 °C. Insets in (b) and (f) are SEM images of the individual  $\text{Mo}_2\text{C}$  crystal grown at 780 and 900 °C with high magnification, respectively.

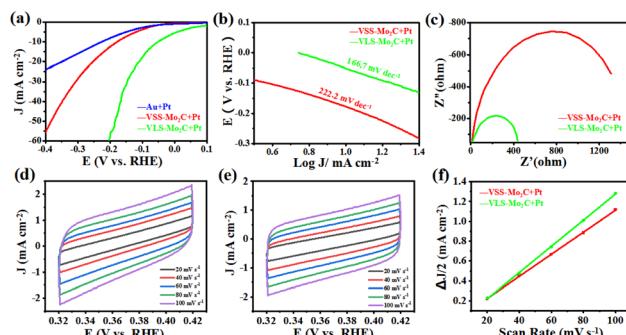


Fig. 4 (a) The LSV curves of Pt/VLS-Mo<sub>2</sub>C, Pt/VSS-Mo<sub>2</sub>C, and Pt/Au with IR correction. (b) The corresponding Tafel slopes. (c) Nyquist plots of Pt/VLS-Mo<sub>2</sub>C and Pt/VSS-Mo<sub>2</sub>C collected at the open-circuit voltage. CV curves at different scan rates from 20 to 100 mV s<sup>-1</sup> of (d) Pt/VLS-Mo<sub>2</sub>C and (e) Pt/VSS-Mo<sub>2</sub>C. (f) The plots of  $\Delta J$  versus scan rates for the Pt/VLS-Mo<sub>2</sub>C and Pt/VSS-Mo<sub>2</sub>C, respectively.

Table 1 Comparison of some molybdenum carbide based electrocatalysts for HER

Samples	$\eta_{10}$ (mV)	Electrolyte	References
Pt/VLS-Mo <sub>2</sub> C	52	1 M KOH	This work
Mo <sub>2</sub> C@NC/Mo <sub>2</sub> C-12	45	1 M KOH	9
C-MoC-0.5	100	1 M KOH	8
Ni/β-Mo <sub>2</sub> C	157	1 M KOH	33
Co-Mo <sub>2</sub> C@NCNT	186	1 M KOH	34
MoC-Mo <sub>2</sub> C-790	98.2	1 M KOH	35
Mo <sub>2</sub> -MoP NPC/CFP-80	146	1 M KOH	36

the Mo precursor. As the growth temperature increased from 850 to 900 °C, the as-grown Mo<sub>2</sub>C crystals gradually adopt block morphology, as shown in Fig. 3e and f. The inset in Fig. 3f is the SEM image of an individual block Mo<sub>2</sub>C crystal.

In order to further demonstrate the advantage of VLS in synthesizing Mo<sub>2</sub>C, the HER catalytic activities of VLS-Mo<sub>2</sub>C and VSS-Mo<sub>2</sub>C were compared. The samples of VLS-Mo<sub>2</sub>C (150 mg per mL Na<sub>2</sub>MoO<sub>4</sub>) and VSS-Mo<sub>2</sub>C [150 mg per mL (NH<sub>4</sub>)<sub>6</sub>Mo<sub>7</sub>O<sub>24</sub>] were synthesized on Au substrates, and both the two kinds of Mo<sub>2</sub>C were loaded with 2 nm Pt for the electrochemical test. The HER catalysis was evaluated in 1.0 M KOH solution using a typical three-electrode system with the studied materials as the working electrodes, Hg/HgO as the reference electrode, and the Pt foil as the counter electrode.

Fig. 4a shows the linear sweep voltammetry (LSV) curves of Pt/VLS-Mo<sub>2</sub>C, Pt/VSS-Mo<sub>2</sub>C, and Pt/Au with a scan rate of 5 mV s<sup>-1</sup>. Compared with the Pt/VSS-Mo<sub>2</sub>C and Pt/Au, the Pt/VLS-Mo<sub>2</sub>C has a lower overpotential of 52 mV *versus* the reversible hydrogen electrode (RHE) at a current density of 10 mA cm<sup>-2</sup>, indicating that the VLS mode can substantially improve the collaborative catalytic performance of Pt/Mo<sub>2</sub>C toward HER in alkaline condition. The derived Tafel slope of Pt/VLS-Mo<sub>2</sub>C and Pt/VSS-Mo<sub>2</sub>C is around 166 and 222 mV dec<sup>-1</sup>, respectively (Fig. 4b), indicating that the hydrogen evolution on both of them undergoes the Volmer mechanism, and water dissociation is the rate-determining step. Critically, a substantially

decreased Tafel slope of Pt/VLS-Mo<sub>2</sub>C revealed that the sluggish water dissociation behavior had improved significantly. In addition, electrochemical impedance spectroscopy (Fig. 4c) showed that Pt/VLS-Mo<sub>2</sub>C possessed a lower charge transfer resistance than Pt/VSS-Mo<sub>2</sub>C. The significantly reduced impedance further suggest that Pt/VLS-Mo<sub>2</sub>C can substantially boost the interfacial electron-transfer kinetics between the Mo<sub>2</sub>C and Au foil, which promotes the HER dynamic process. The electrochemical surface areas of Pt/Mo<sub>2</sub>C crystals were further estimated by deriving the electrochemical double layer capacitance ( $C_{dl}$ ) from the cyclic voltammetry studies, as shown in Fig. 4d-f. The Pt/VLS-Mo<sub>2</sub>C was found to have a larger  $C_{dl}$  of 13.2 mF cm<sup>-2</sup> than Pt/VSS-Mo<sub>2</sub>C (11.1 mF cm<sup>-2</sup>), indicating that the VLS mode can increase the electrochemical surface areas of the as-grown Mo<sub>2</sub>C crystals.

We compared the Pt/VLS-Mo<sub>2</sub>C over the state-of-the-art of electrocatalysts for HER, as shown in Table 1. We believe that the VLS method could offer new insights into the synthetic approaches for Mo<sub>2</sub>C and provide new strategies for constructing metal-loading catalysts with high HER catalytic activity.

## Conclusion

In summary, we demonstrated the VLS growth of  $\alpha$ -Mo<sub>2</sub>C micron flowers, which were realized by using liquid precursor for the first time. The morphology and density of the Mo<sub>2</sub>C crystals could be controlled by tuning the growth temperature and concentration of Na<sub>2</sub>MoO<sub>4</sub> aqueous solution. The unique flower-like structure produces a high specific surface area and abundant surface sites on the surface, increasing the Pt loading and enhancing the collaborative catalysis. The comparison between Pt/VLS-Mo<sub>2</sub>C and Pt/VSS-Mo<sub>2</sub>C in terms of HER catalytic activities further demonstrated the advantage of VLS in synthesizing Mo<sub>2</sub>C crystals. Our study not only offers new insights into the synthetic approaches for Mo<sub>2</sub>C but also provides a new strategy for constructing metal-loading catalysts with high catalytic activity.

## Author contributions

Bin Wang designed and conducted the VLS growth and analyzed the data. Yipeng Zang performed the HER of the materials. All the authors discussed and commented on the manuscript.

## Conflicts of interest

The authors declare that they have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper.

## Acknowledgements

This work was financially supported by the National Natural Science Foundation of China (No. 21688102 and No. 21825203), the National Key R&D Program of China (No. 2016YFA0200200),



and the Strategic Priority Research Program of the Chinese Academy of Sciences (Grant No. XDB17020000).

## Notes and references

- 1 A. W. H. Stefan, W. G. Robert, S. E. Daan and H. B. Johannes, *ACS Catal.*, 2013, **3**, 2837–2844.
- 2 M. Q. Zeng, Y. X. Chen, J. X. Li, H. F. Xue, R. G. Mendes, J. X. Liu, T. Zhang, M. H. Rümmeli and L. Fu, *Nano Energy*, 2017, **33**, 356–362.
- 3 M. R. Lukatskaya, O. Mashtalir, C. E. Ren, Y. Dall'Agnese, P. Rozier, P. L. Taberna, M. Naguib, P. Simon, M. W. Barsoum and Y. Gogotsi, *Science*, 2013, **341**, 1502–1505.
- 4 K. Ba, G. L. Wang, T. Ye, X. R. Wang, Y. Y. Sun, H. Q. Liu, A. Q. Hu, Z. Y. Li and Z. Z. Sun, *ACS Catal.*, 2020, **10**, 7864–7870.
- 5 W. Y. Sun, X. Q. Wang, J. Q. Feng, T. Li, Y. H. Huan, J. B. Qiao, L. He and D. L. Ma, *Nanotechnol.*, 2019, **30**, 385601–385608.
- 6 J. S. Li, Y. Wang, C. H. Liu, S. L. Li, Y. G. Wang, L. Z. Dong, Z. H. Dai, Y. F. Li and Y. Q. Lan, *Nat. Commun.*, 2016, **7**, 11204.
- 7 D. Geng, X. Zhao, Z. Chen, W. Sun, W. Fu, J. Chen, W. Liu, W. Zhou and K. P. Loh, *Adv. Mater.*, 2017, **29**, 1700072.
- 8 Y. C. Liu, B. B. Huang, X. Hu and Z. L. Xie, *Int. J. Hydrogen Energy*, 2019, **44**(7), 3702–3710.
- 9 X. F. Zhang, T. Lei, M. Xia, Q. H. Wei and Z. L. Xie, *Dalton Trans.*, 2023, **52**, 6267–6272.
- 10 Y. Z. Ge, X. T. Qin, A. W. Li, Y. C. Deng, L. L. Lin, M. T. Zhang, Q. L. Yu, S. W. Li, M. Peng, Y. Xu, X. Y. Zhao, M. Q. Xu, W. Zhou, S. Y. Yao and D. Ma, *J. Am. Chem. Soc.*, 2021, **143**(2), 628–633.
- 11 L. L. Lin, Q. L. Yu, M. Peng, A. W. Li, S. Y. Yao, S. H. Tian, X. Liu, A. Li, Z. Jiang, R. Gao, X. D. Han, Y. W. Li, X. D. Wen, W. Zhou and D. Ma, *J. Am. Chem. Soc.*, 2021, **143**, 309–317.
- 12 X. Zhang, M. T. Zhang, Y. C. Deng, M. Q. Xu, L. C. Artiglia, W. Wen, R. Gao, B. B. Chen, S. Y. Yao, X. C. Zhang, M. Peng, J. Yan, A. W. Li, Z. Jiang, X. Y. Gao, S. F. Cao, C. Yang, A. J. Krop, J. N. Shi, J. L. Xie, M. S. Bi, J. A. Bokhoven, Y. W. Li, X. D. Wen, M. Flytzani-Stephanopoulos, C. Shi, W. Zhou and D. Ma, *Nature*, 2021, **589**, 396–401.
- 13 S. Posada-Pérez, R. A. Gutiérrez, Z. J. Zuo, P. J. Ramírez, F. Viñes, P. Liu, F. Illas and J. A. Rodriguez, *Catal. Sci. Technol.*, 2017, **7**, 5332–5342.
- 14 J. B. Wu, L. K. Xiong, B. T. Zhao, M. L. Liu and L. Huang, *Small Methods*, 2020, **4**, 1900540.
- 15 X. Y. Li, D. Ma, L. M. Chen and X. H. Bao, *Catal. Lett.*, 2007, **116**, 63–69.
- 16 T. C. Xiao, A. P. E. York, H. Al-Megren, C. V. Williams, H. T. Wang and M. L. H. Green, *J. Catal.*, 2001, **202**(1), 100–109.
- 17 T. C. Xiao, A. P. E. York, V. C. Williams, H. Al-Megren, A. Hanif, X. Y. Zhou and M. L. H. Green, *Chem. Mater.*, 2000, **12**(12), 3896–3905.
- 18 S. R. Vallance, S. Kingman and D. H. Gregory, *Chem. Commun.*, 2007, **7**, 742–744.
- 19 M. Saito and R. B. Anderson, *J. Catal.*, 1980, **63**, 438–446.
- 20 L. Volpe and M. Boudart, *J. Solid State Chem.*, 1985, **59**, 348–356.
- 21 H. Liu, G. P. Qi, C. S. Tang, M. L. Chen, Y. Chen, Z. W. Shu, H. Y. Xiang, Y. Y. Jin, S. S. Wang, H. M. Li, M. Ouzounian, T. S. Hu, H. G. Duan, S. S. Li, Z. Han and S. Liu, *ACS Appl. Mater. Interfaces*, 2020, **12**, 13174–13181.
- 22 S. M. Feng, J. Y. Tan, S. L. Zhao, S. Q. Zhang, U. Khan, L. Tang, X. L. Zou, J. H. Lin, H. M. Cheng and B. L. Liu, *Small*, 2020, **2003357**, 1–9.
- 23 S. S. Li, Y. C. Li, X. Y. Liu, Z. H. Hu, J. Wu, H. Nakajima, S. Liu, T. Okazaki, W. Chen, T. Minari, Y. Sakuma, K. Tsukagoshi, K. Suenaga, T. Taniguchi and M. Osada, *Nanoscale*, 2019, **11**, 16122–16129.
- 24 S. S. Li, Y. C. Lin, W. Zhao, J. Wu, Z. Wang, Z. H. Hu, Y. D. Shen, D. M. Tang, J. Y. Wang, Q. Zhang, H. Zhu, L. Q. Chu, W. J. Zhao, C. Liu, Z. P. Sun, T. Taniguchi, M. Osada, W. Chen, Q. H. Xu, A. T. S. Wee, K. Suenaga, F. Ding and G. Eda, *Nat. Mater.*, 2018, **17**, 535–542.
- 25 M. Zeng and L. Fu, *Acc. Chem. Res.*, 2018, **51**, 2839–2847.
- 26 H. Cheng, L. X. Ding, G. F. Chen, L. Zhang, J. Xue and H. Wang, *Adv. Mater.*, 2018, **30**, 1803694.
- 27 J. Halim, S. Kota, M. R. Lukatskaya, M. Naguib, M. Q. Zhao, E. J. Moon, J. Pitock, J. Nanda, S. J. May, Y. Gogotsi and M. W. Barsoum, *Adv. Funct. Mater.*, 2016, **26**, 3118–3127.
- 28 Q. Gao, X. Y. Zhao, Y. Xiao, D. Zhao and M. H. Cao, *Nanoscale*, 2014, **6**, 6151–6157.
- 29 R. R. Li, S. G. Wang, W. Wang and M. H. Cao, *Phys. Chem. Chem. Phys.*, 2015, **17**, 24803–24809.
- 30 D. C. Geng, X. X. Zhao, L. J. Li, P. Song, B. B. Tian, W. Liu, J. Y. Chen, D. Shi, M. Lin, W. Zhou and K. P. Loh, *2D Mater.*, 2017, **4**, 011012.
- 31 T. Li, W. Luo, H. Kitadai, X. Wang and X. Ling, *Adv. Mater.*, 2019, **31**, 1807160.
- 32 T. C. Xiao, A. P. E. York, H. Al-Megren, C. V. Williams, H. T. Wang and M. L. H. Green, *J. Catal.*, 2001, **202**, 100–109.
- 33 T. Ouyang, A. N. Chen, Z. Z. He, Z. Q. Liu and Y. Tong, *Chem. Commun.*, 2018, **54**, 9901–9904.
- 34 L. Ai, J. Su, M. Wang and J. Jiang, *ACS Sustainable Chem. Eng.*, 2018, **6**, 9912–9920.
- 35 W. Liu, X. T. Wang, F. Wang, K. F. Du, Z. F. Zhang, Y. Z. Guo, H. Y. Yin and D. H. Wang, *Nat. Commun.*, 2021, **12**, 6776.
- 36 T. T. Liu, H. Liu, X. J. Wu, Y. L. Niu, B. M. Feng, W. Li, W. H. Hu and C. M. Li, *Electrochim. Acta*, 2018, **281**, 710–716.

