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The effect of cetyltrimethylammonium bromide (CTAB) addition on green synthesis of porous N-doped TiO₂ for photoreduction of heavy metal ion Cr(vi)

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In this study, porous TiO₂ photocatalysts modified by nitrogen (NCT) were successfully synthesized using a combination of green synthesis methods by utilizing *Aloe vera* (L.) Burm. f. peel and hydrothermal method. In addition, TiO₂ was modified by increasing the active surface area using Cetyltrimethylammonium Bromide (CTAB). The X-ray Diffraction (XRD) results indicated that the anatase phase was formed. The result of the Diffuse Reflectance Spectroscopy UV-Vis (DRS UV-Vis) using the Tauc-plot method showed that all porous N-doped TiO₂ samples experienced a decrease in the energy gap. This indicates the successful modification of TiO₂ by nitrogen, as confirmed by the Fourier Transform Infra-Red (FTIR) result. Field Emission Scanning Electron Microscopy (FESEM) result showed that the synthesized TiO₂ had a spherical morphology of 10–30 nm diameter. The Braunauer, Emmett, and Teller (BET) result indicated that the type IV isotherm curve with a mesoporous structure was formed. The NCT0.75 sample had a surface area and pore size of 95.02 m² g⁻¹ and 8.021 nm, respectively, while the NTi0.75 sample had a surface area and pore size of 90.97 m² g⁻¹ and 5.161 nm, respectively. The photocatalytic activity of the porous N-doped TiO₂ was tested on photoreduction of metal pollutant model Cr(vi). The result demonstrated that the NCT0.75 sample had the most optimal photocatalytic activity by reducing 89.42% of Cr(vi) metal ions.

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Introduction

As a developing country, Indonesia has begun to show high economic growth, particularly in the industrial sector. The rapid development of industry harms the environment, such as producing and discharging hazardous waste into the environment without prior processing.¹ Heavy metal waste, such as lead (Pb), cadmium (Cd), mercury (Hg), and chromium (Cr), are produced mainly through industrial activities.² According to the World Health Organization (WHO), chromium(vi) waste is one of the carcinogenic pollutants to humans.³ Cr(vi) is mostly produced from industrial activities, such as metallurgy, the leather industry, the electroplating industry, and the wood industry. This condition must be overcome immediately. For this purpose, one way is to change Cr(vi) to Cr(III) to reduce the level of poison.⁴

Photocatalytic is an effective, non-toxic, economical, and easy method to convert Cr(vi) to Cr(III). The photocatalytic process requires a catalyst, such as semiconductor catalysts,

including TiO₂, ZnO, ZnS, ZrO₂, CdS, and WO₃. Due to its electronic and optical properties, high photocatalytic activity, and non-toxicity, TiO₂ is considered an ideal semiconductor catalyst.⁵ Pure TiO₂ with an anatase phase has a band gap value of 3.2 eV, which is limited only under UV light. Several attempts have been made to expand the working ability of TiO₂ to the visible light region, namely through metal doping, non-metal doping, and sensitization. Li *et al.* (2020) showed that nitrogen is the most effective dopant used because it can modify the electronic structure of TiO₂ by forming a new band (N 2p orbital) above the valence band (O 2p orbital).^{6,7} In addition, efforts to increase photocatalytic activity can also be conducted by expanding the surface of the photocatalyst with the formation of pores.⁸

TiO₂ nanoparticles can be synthesized using chemical and physical methods, such as solvothermal, sol-gel, hydrothermal, sonochemical, microwave, and ultrasonic irradiation. According to Byrappa and Adschi (2007) and Hayashi and Hakuta (2010), hydrothermal synthesis is usually understood to involve heterogeneous reactions occurring in aqueous or non-aqueous fluids that result in crystallization above room temperature and at high pressure (>1 atm). The majority of scientists have used the word “solvothermal” to describe chemical reactions that happen in non-aqueous solvents. The closed stainless steel

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autoclave used for hydrothermal synthesis often has a Teflon coating to resist the high heat and pressures involved in this process.⁹ So this method requires high capital requirements for instrumentation, and the inability to monitor crystal growth, this method can only be carried out in supercritical solvent conditions, and uses non-aqueous solvents with very high boiling points.¹⁰

Physical methods include thermal decomposition, laser irradiation, and electrolysis while chemical methods use chemicals such as sodium borohydride or sodium citrate as reducing agents. Physical methods typically use highly intensive costly equipment that involves vacuum and chemical methods are considered the most commonly used technique to synthesize nanoparticles. Preparation through chemical and physical methods can lead to environmental problems upon discharge due to the usage of dangerous chemicals.¹¹ So these methods not only require high energy and high pressure, but also they are expensive. Moreover, the chemical processes involve reducing and stabilizing agents producing by-products that are not environmentally friendly.¹² Therefore, the nanoparticle biosynthesis method is used because of being environmentally friendly without the use of complex equipment and techniques and also high temperatures and pressures.¹³

Medicinal plant *Aloe vera* (L.) Burm. f. can be selected as a reducing and stabilizing agent in the biosynthesis of N-doped TiO₂ because it contains secondary metabolites, such as saponins, flavonoids, and tannins, acting as biogenic capping agents.^{14,15} A capping agent is a functional compound that maintains the shape of the particles by controlling the direction of crystal growth. Crystal growth can run fast which causes the resulting particle size to be large. Specifically, the capping agent will be in a particular crystal plane to lower the surface energy to direct the process nucleation forms stable TiO₂.¹⁷

The application of N-doped TiO₂ with high crystallinity and specific surface area is very important to increase photocatalytic efficiency. One way to develop a material with a high surface area is to introduce a pore-forming agent to the synthesis of the material. The synthesis of porous N-doped TiO₂ using CTAB as a pore-forming agent was performed by Toe *et al.* (2018), leading to the production of a porous material with a high, uniform, and homogeneous specific surface area, thereby increasing photocatalytic activity.¹⁶ Therefore, in this study, CTAB was used as a pore-forming agent in synthesizing porous N-doped TiO₂ and utilizing *Aloe vera* (L.) Burm. f. bark extract as a natural capping agent using a combination of biosynthetic and hydrothermal methods. The resulting porous N-doped TiO₂ was applied to reduce metal ions Cr(vi).

Experimental

Materials

The materials used in this study were TiCl₄ (Merck) as TiO₂ precursor, *Aloe vera* (L.) Burm. f. (*Aloe vera*), NH₄OH 25% as nitrogen source, commercial cetyltrimethylammonium bromide (CTAB) (Merck) as porous forming agent, deionized water, ethanol (p.a. Merck), aquadest, NaOH (Merck), K₂Cr₂O₇ (Merck) for pollutant model, H₂SO₄ (p.a. Merck), and 1,5-

diphenylcarbazine (Merck) as complexing agent. All materials were used as received. The equipment used includes glassware, funnels, Whatman No. 42 filter paper, magnetic stirrer, hot plate, cylinder vessel, autoclave, furnace, centrifuge, oven, analytical balance, pH meter, aluminium foil, and light sources (VISALUX LED lamps 24 watts). The instruments used for sample characterization were X-ray Diffraction (XRD-XPRT PRO PANalytical pw 30/40), Diffuse Reflectance Spectroscopy UV-Vis (DRS UV-Vis-SPECORD 210 Plus), Fourier Transform Infra-Red (FTIR-PerkinElmer Version 10.6.1), Braunauer, Emmett, and Teller analysis (BET-Quantachrome Nova 4200e), Field Emission Scanning Electron Microscopy (FESEM) and Electron Dispersive X-ray (EDX-FEI Quanta FEG 650), and Ultraviolet-visible (UV-Vis) Spectrophotometry (SHIMADZU CORP 04843).

Preparation of *Aloe vera* (L.) Burm. f. extract

To prepare the *Aloe vera* samples, the peel was taken and separated from the gel 10 g of fresh sample was washed and mashed. Then the sample was boiled with deionized water (100 mL) and stirred for 1 h at 70 °C. The bark extract was filtered using the Whatman filter paper.¹⁸

Synthesis of porous TiO₂ nanoparticles

5 mL of titanium chloride (TiCl₄ 1 M) solution was put into a flat bottom flask and stirred at a constant speed (1500 rpm) in a closed state. Then, 50 mL of deionized water and *Aloe vera* (L.) Burm. f. peel extract with a volume concentration of 10% v/v were added dropwise under constant stirring. After adding the extract, 0.925 g of CTAB and 1 M NaOH were added to obtain pH ~ 7. After stirring for 24 h, the formed nanoparticles were centrifuged (10 000 rpm) and washed with ethanol and distilled water three times each to remove impurities. Then the obtained precipitate was dried at 105 °C for 10 h and calcined at 500 °C for 1 h. The same treatment was also conducted on the control, namely the synthesis product, without adding CTAB.¹⁶ The obtained porous TiO₂ nanoparticle from this initial step will use as the substrate for the N-doping process.

Synthesis of porous N-doped TiO₂

The porous N-doped TiO₂ was prepared using the hydrothermal method. A total of 0.2 grams of porous TiO₂ was added to a mixture of 8 mL deionized water and 12 mL ethanol in a 50 mL cylindrical vessel. After adding 0.25 mL of 25% NH₄OH, the solution was stirred for 2 h. The obtained mixture was put into a 50 mL Teflon-coated autoclave at 180 °C for 12 h. Afterward, the samples were centrifuged (10 000 rpm), repeatedly washed with ethanol and distilled water, and dried in an oven at 105 °C for 10 h. The resulting product was labelled NCT0.25. The same steps were repeated for different volumes of 25% NH₄OH, namely 0.5 mL, 0.75 mL, and 1 mL, with the product labels NCT0.5, NCT0.75, and NCT1 (see Table 1). TiO₂ control samples were synthesized without the addition of CTAB (labelled as NTi) and NH₄OH (labelled as CTO).¹⁹



Table 1 Variation of N : Ti mole ratio of samples

No.	Sample	Sample code	Mole ratio N : Ti
1	Non-doped porous TiO ₂	CTO	—
2	N-Doped TiO ₂ using CTAB	NCT0.25	1 : 1
3	N-Doped TiO ₂ using CTAB	NCT0.5	2 : 1
4	N-Doped TiO ₂ using CTAB	NCT0.75	3 : 1
5	N-Doped TiO ₂ using CTAB	NCT1	4 : 1
6	N-Doped TiO ₂ without CTAB	NTi0.75	3 : 1

Characterization of porous TiO₂ nanoparticles

The samples were characterized using XRD to detect the crystal phase of the sample. Then the refinement process was performed on XRD using the Rietica program with the Le Bail method to obtain crystallographic data of the synthesized material. The optical properties of the samples were studied by DRS UV-Vis. FTIR was employed to determine the chemical structures of porous N-doped TiO₂ samples. The surface morphology and composition of the samples were evaluated by FESEM and EDX, respectively. In addition, BET analysis was utilized to determine the specific surface area, pore diameter, and pore volume of the samples.

Photocatalytic activity test

0.01 g of K₂Cr₂O₇ was dissolved in 1 L of solution. H₂SO₄ 6 N was added until the pH reached 2 and stirred for 30 minutes. After that, 0.05 g porous N-doped TiO₂ sample was inserted into 10 mL of that solution. The mixture was stirred for 5 min until N-doped TiO₂ was dispersed homogeneously. Then, the mixture was continue stirred in the dark for 210 min to reach adsorption/desorption equilibrium. After keeping for 210 min, the mixture was placed in a bright place irradiated by a 24 watt Light Emitting Diode (LED) lamp for 240 min. The mixture was put into a test tube and centrifuged. After separating the filtrate, 10 drops of 0.5% w/v 1,5-diphenylcarbazide as coloring agent was added forming complex solution. The absorbance of this complex solution was measured by UV-Vis spectrophotometer at a 543 nm wavelength. The same steps were repeated for NCT0.25, NCT0.5, NCT0.75, and NCT1. Porous TiO₂ without adding NH₄OH (CTO) and N-doped TiO₂ without adding CTAB (NTi0.75) samples were used as control.²⁰ The efficiency of Cr(vi) reduction can be calculated using the following formula:

$$\% \text{ Reduction} = \frac{A_i - A_f}{A_i} \times 100$$

where A_i and A_f are the absorbances of Cr(vi) before and after irradiation, respectively. Each experiment was carried out twice, with the average data reported.

Result and discussion

Porous N-doped TiO₂ nanoparticles

Porous N-doped TiO₂ nanoparticles were synthesized using a combination of green synthesis methods (using *Aloe vera* as a natural capping agent) and hydrothermal method in the TiO₂

doping process. Rahim *et al.* (2020) showed that the results of phytochemical tests on the skin of *Aloe vera* confirmed the presence of secondary metabolites, such as tannins, saponins, flavonoids, and anthraquinones.²¹ *Aloe vera* peel which acts as a natural capping agent, contains compounds functioning as strong chelating, reducing, and stabilizing agents in maintaining the shape and size of nanoparticles and preventing agglomeration.¹⁵ Flavonoid and tannin compounds found in *Aloe vera* skin can bind Ti metal due to possessing hydroxyl (–OH) and carbonyl (C=O) groups. In the ionic state, Ti⁴⁺ repulsion occurs due to a similar charge. After being reduced to Ti⁰, Ti atoms can interact with each other through metallic bonds to form nano-sized clusters. The neutral charge of the Ti atom is due to the electron donation to the Ti⁴⁺ ion by the hydroxyl and carbonyl functional groups to produce Ti nanoparticles and to prevent agglomeration as a stabilizer.^{22,23} The mechanism of reduction of TiO₂ nanoparticles by plant extracts can be seen in Fig. 1.

Fig. 2 shows that all the synthesis results obtained are in powder form. The influence of the N/Ti mole ratio can be seen from the color of the TiO₂ powder produced, namely pure white powder for undoped TiO₂ (CTO) and yellowish-white for N-doped TiO₂ (NCT) samples. Samples that are yellowish-white indicate that doping using nitrogen has been carried out successfully.¹⁶ Meanwhile, the results of previous studies reported that the synthesis of C–TiO₂ with carbon dopant obtained a white powder before calcination and a grayish-white powder after the calcination process.²⁴ Whereas the synthesis of Fe–TiO₂ with Fe dopant shows the brown color from Fe-doped TiO₂.²⁵ So that it can be confirmed that it is a nitrogen dopant that has been induced into TiO₂.

FESEM-EDX analysis

Based on Fig. 3, the formation of particles in the N-doped TiO₂ sample with the addition of CTAB (NCT0.75) displays a more perfect and homogeneous spheric/spherical morphology compared to the N-doped TiO₂ sample without CTAB (NTi0.75). The NCT0.75 sample had a dominant particle size distribution in the range of 18–20 nm in addition to pores due to the addition of CTAB. The porous structure is caused by the formation of CTAB micelles, which function as pore molds assisting the production of surfaces with porous structure after the calcination process.

The role of CTAB as a capping agent is able to control particle size during the synthesis process so that the particle size becomes smaller and more uniform.²⁶ Meanwhile, NTi0.75

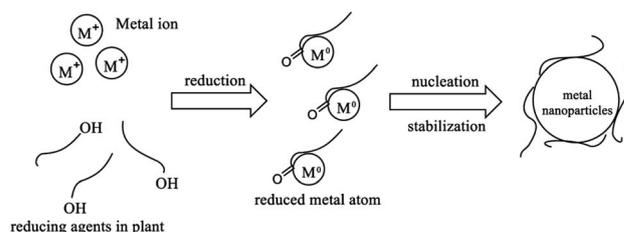


Fig. 1 Mechanism of metal nanoparticles synthesis by plant extracts.²³





Fig. 2 Synthesized porous TiO₂ and N-doped porous TiO₂ nanoparticle samples: (a) CTO, (b) NCT0.25, (c) NCT0.5, (d) NCT0.75, and (e) NCT1.

showed that the particle shape was not entirely spherical, because the rod-shaped particles were found. Also, the dominant particle size distribution was in the range of 10–14 nm. Additionally, it can be seen that the NTi0.75 sample had a denser structure. This is due to the absence of CTAB as a pore-forming agent and particle size-controlling agent.²⁷

EDX characterization was employed to determine the elemental composition of the samples, as shown in Fig. 4. Fig. 4 illustrates that the NCT0.75 and NTi0.75 samples had sharp and strong peaks in titanium (Ti) and oxygen (O). A nitrogen (N) peak was not detected in both samples because the detector was not sensitive enough to analyze elements with a small atomic number ($Z < 11$), such as nitrogen and boron. The EDX detector able to determine elemental nitrogen is the ultra-thin window (UTW) which has a high sensitivity.²⁸

On the other hand, the absence of a nitrogen peak may be caused by the trace amount of added nitrogen as a dopant, so the instrument cannot detect the element. To confirm the presence of nitrogen-doped on TiO₂, further characterization was conducted. For instance, the XRD data was utilized for the refinement process so that a peak shift could be seen in the presence of nitrogen doping on TiO₂.

XRD analysis

The crystal phase, crystal size, and crystallinity of the synthesized compounds were investigated by XRD.²⁹ The XRD patterns of the TiO₂ samples with the addition of nitrogen-doped CTAB are shown in Fig. 5. It can be seen that the standard anatase TiO₂ had Bragg diffraction peaks with the 2θ values of 25.3°,

37.87°, 48.05°, 53.93°, and 55.10° indexed in planes (101), (004), (200), (105), and (211).

The results of this study indicate that there is no significant difference between the obtained 2θ values and the standard data, both showing anatase crystals. However, it was observed that there is a phase transformation from anatase to brookite in NCT0.25, NCT0.5, and NCT0.75. This is indicated by a characteristic peak at $2\theta = 30.8^\circ$, which can be originated from the brookite crystal surface indexed in the plane (121), based on ICSD standard no. 154605.

This result indicates that adding an appropriate nitrogen amount to the hydrothermal solution can cause the change of crystal phase from the anatase to an anatase–brookite mixed phase, better known as a heterostructured material, while the excess addition of nitrogen produces anatase TiO₂ products with relatively low crystallinity, it can be seen from the NCT1 diffractogram that there is no brookite peak at $2\theta = 30.8^\circ$.^{30,31} The formation of this heterojunction mixture enhances the photoreduction results, especially in the NCT0.75 sample, which can be considered the most optimum sample compared to others.

Nitrogen deposition has the effect of reducing band gap energy and making the material more active in visible light. The XRD diffractogram in Fig. 5 displays the typical TiO₂ peak appearance. The successful impregnation of TiO₂ or N-TiO₂ serves as evidence of this. The anatase structure cannot be destroyed by doping TiO₂, as the N-TiO₂ diffractogram shows peaks that are comparable to TiO₂. But according to earlier studies, the doping procedure might render a crystal lattice ineffective, and its result made N-doped TiO₂ more photocatalytically active due to the decreased band gap energy.³²

The crystal size of the synthesized material can be calculated from the Full Width at Half Maximum (FWHM) value at peak (101) using the Debye–Scherrer equation. From the calculation results, the crystal size of the control TiO₂ was 7.61 nm. The crystal size increased to 8.37 nm in the NCT0.25 sample by doping due to the larger atomic radius of nitrogen (N^{3-} 1.46 Å) than that of oxygen (O^{2-} 1.40 Å). Therefore, the crystal size increased by nitrogen doping. This can be explained by the fact that nitrogen doping can sharpen the XRD peak by increasing the crystal growth rate and the crystal size.³³

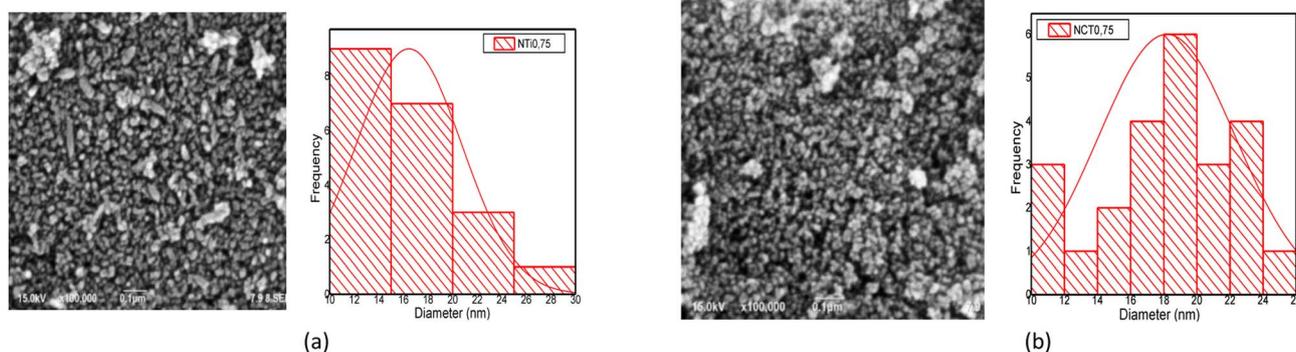


Fig. 3 FESEM images and particle size distribution of samples: (a) NTi0.75 and (b) NCT0.75.



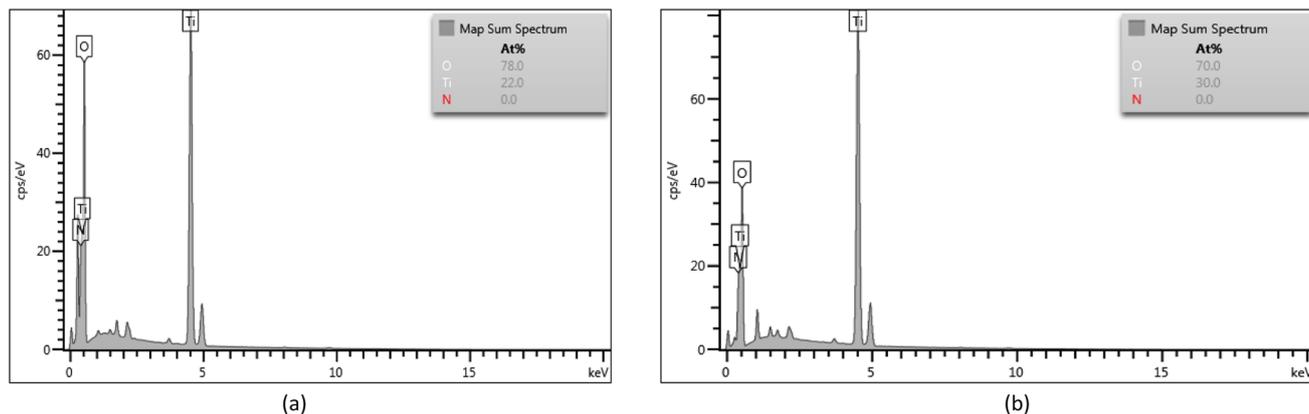


Fig. 4 EDX spectrum of samples: (a) NCT0.75, (b) NTi0.75.

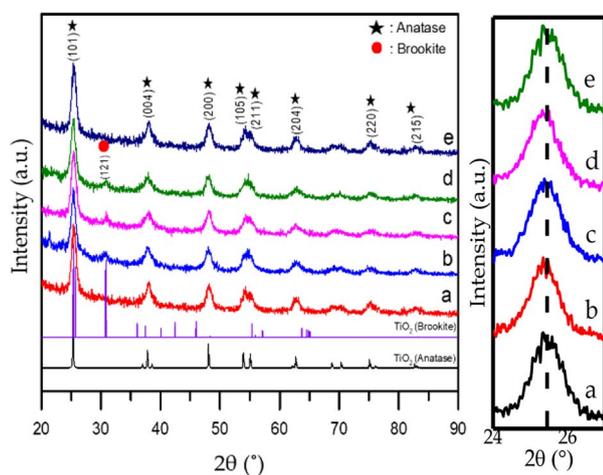


Fig. 5 XRD diffraction patterns of samples: (a) CTO, (b) NCT0.25, (c) NCT0.5, (d) NCT0.75, and (e) NCT1.

Further analysis was performed by the refinement process using the Rietica program with the Le Bail method to obtain crystallographic data from the synthesized TiO_2 and determine the effect of nitrogen dopant on TiO_2 crystallographic data (Fig. 6). Refinement was conducted using the TiO_2 standard anatase phase ICSD no. 9855 with a tetragonal structure, space group $I4_1/amdS$, length a and b 3.8040 Å, length c 9.614 Å, and cell volume 139.12 Å³. The refined results of nitrogen-doped porous TiO_2 with variations in dopant volume of nitrogen (0, 0.25, 0.5, 0.75, and 1 mL) are shown in Table 2. The diffraction data of the nitrogen-doped porous TiO_2 sample is in good agreement with the anatase standard. According to Andrieux *et al.* (2018), smoothing success is indicated by obtaining an R -profile (R_p) of less than 15%, an R -weight profile (R_{wp}) of less than 20%, and a goodness of fit (X^2) of less than 5.³⁴

Table 2 demonstrates that undoped TiO_2 with N-doped TiO_2 does not change its structure, such as space group, crystal lattice, and asymmetry. The magnification of the peak with the highest intensity indicates that the peak shift occurs due to the lattice tension on the crystal size. It can be seen that adding nitrogen dopants causes an increase in the values of lattice parameters: a ,

b , and c , as well as cell volume. The increase in cell volume was caused by the doping of nitrogen (1.46 Å) which has a larger atomic radius than oxygen (1.40 Å). So, the position of the oxygen atom can be occupied by nitrogen increasing the crystal lattice and hence the cell volume. The increase in cell volume after doping with nitrogen indicates that nitrogen was doped on the TiO_2 lattice. It also confirms the peak shift in the XRD pattern toward a smaller 2θ .^{33,35}

DRS UV-Vis analysis

By UV-Vis DRS characterization, the absorption of light energy and the optical properties of nitrogen-doped TiO_2 nanoparticles were investigated. Fig. 7 displays the absorption intensity of the control TiO_2 sample (CTO) occurring in the UV light region at a wavelength <400 nm. However, all nitrogen-doped TiO_2 catalysts showed a shift in absorption intensity to the visible region of ~410 nm, confirming the successful doping. This result is supported by a report from Dong *et al.* (2018) stating that adding nitrogen dopant can shift the absorption intensity of TiO_2 to the visible light region.³⁶

Based on the results of DRS UV-Vis, the band gap energy for each sample can be calculated using the Tauc plot method, as shown in Fig. 8. In this study, the control TiO_2 sample (CTO) had an energy gap of 3.2 eV. After doping using nitrogen, there was a decrease in the energy gap for samples NCT0.25, NCT0.5, and NCT0.75 by 3.16 eV, 3.15 eV, and 3.14 eV, respectively. However, the NCT1 sample experienced an increase in the energy gap to 3.16 eV. It is possible that the NCT1 sample had a change in crystal structure and composition due to the influence of adding excess nitrogen dopant.¹⁶

The most optimum sample synthesized in this study was NCT0.75 with the band gap energy of 3.14 eV, the lowest achieved value. The decrease in the gap energy as the result of nitrogen doping proves the successful doping of nitrogen on TiO_2 . It was confirmed that nitrogen dopant contributes to the narrowing of the band gap of the photocatalyst by forming a new N 2p band between the O 2p and Ti 3d orbitals in the valence band of the TiO_2 electronic structure. Photons can trigger the transfer of electrons from the valence band to the conduction band by



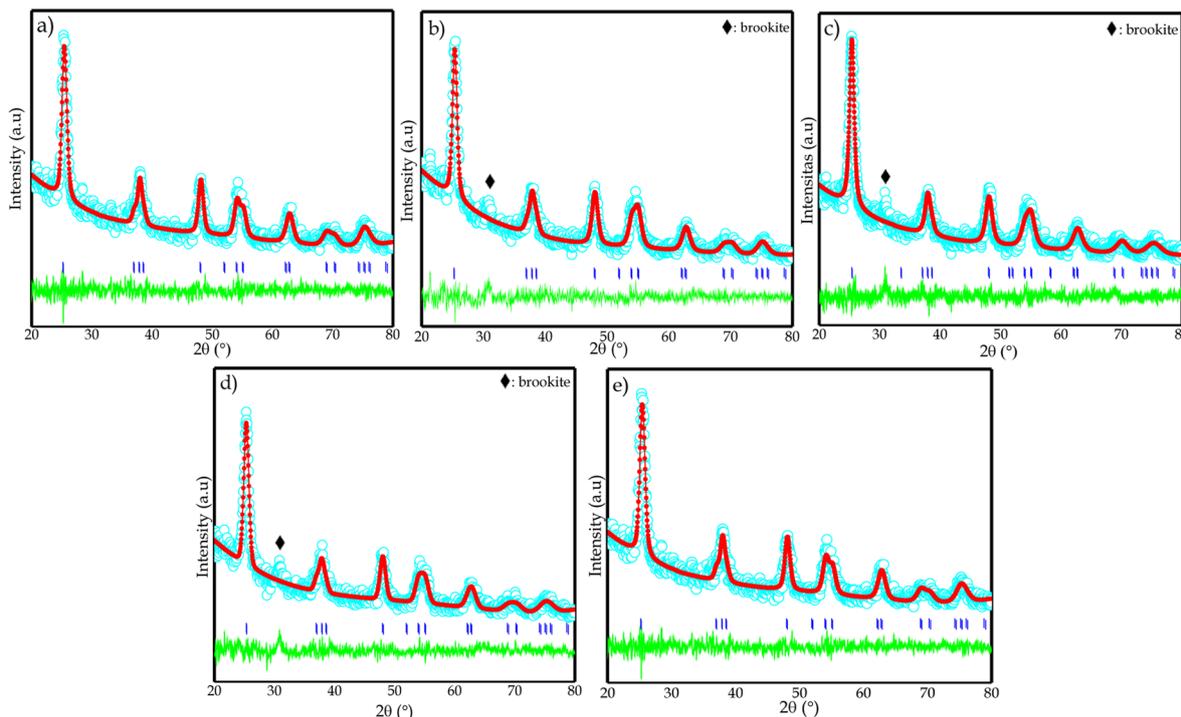


Fig. 6 Le Bail plot of samples: (a) CTO, (b) NCT0.25, (c) NCT0.5, (d) NCT0.75, and (e) NCT1 based on XRD data with space group $I4_1/amdS$.

absorbing energy in the visible region corresponding to the lower band gap energy.⁷

FTIR analysis

FTIR characterization was employed to detect bond vibrations or interactions in the formation of porous N-doped TiO_2 . Measurements were made in the range of $400\text{--}4000\text{ cm}^{-1}$. The FTIR spectra of undoped TiO_2 and nitrogen-doped TiO_2 samples are shown in Fig. 9. Also, the identification of the FTIR vibration types is tabulated in Table 3. The spectra of the N-doped TiO_2 sample without CTAB (NTi0.75) had the same pattern as those of the TiO_2 and N-doped TiO_2 using CTAB (CTO and NCT0.75). This is because all samples underwent a calcination process, so no peaks indicated the presence of CTAB in the samples. The calcination process causes the breaking of many bonds of organic compounds, such as CTAB, followed by decomposition.³⁷

Nitrogen-doped TiO_2 samples (NCT0.75 and NTi0.75) showed absorption peaks in the region of $1530\text{--}1640\text{ cm}^{-1}$ while the TiO_2 spectra without doping (CTO) only exhibited absorption peaks in the region of 1629 cm^{-1} . The absorption peaks at 1539 cm^{-1} and 1535 cm^{-1} corresponded Ti–N bonds caused by NH_2 vibrations indicating that nitrogen was doped to the titanium metal center. In Fig. 9a, the FTIR spectra indicate a wide band at wavenumbers $3350\text{--}3750\text{ cm}^{-1}$, which is an O–H strain vibration adsorbed on the surface. The absorption peak also appears at wave number 1629 cm^{-1} , which is a bending vibration of dissociated O–H or water molecules adsorbed on the surface of TiO_2 and the presence of N–H bonds.³⁸ Fig. 9b and c illustrate that the resulting absorption bands are smaller due to the evaporation of the hydroxyl groups and water during the calcination and hydrothermal processes.³⁹ The TiO_2 spectra had broad

Table 2 Unit cell parameters of refined TiO_2 and N-doped TiO_2 compounds in the $I4_1/amdS$ group space

Parameters	CTO	NCT0.25	NCT0.5	NCT0.75	NCT1
Space group	$I4_1/amdS$	$I4_1/amdS$	$I4_1/amdS$	$I4_1/amdS$	$I4_1/amdS$
Crystal system	Tetragonal	Tetragonal	Tetragonal	Tetragonal	Tetragonal
Asymmetric unit (Z)	4	4	4	4	4
a (Å)	3.7786	3.7866	3.8024	3.7915	3.7791
b (Å)	3.7786	3.7866	3.8024	3.7915	3.7791
c (Å)	9.4693	9.4957	9.5385	9.5104	9.4703
V (Å ³)	135.2034	136.1571	137.9071	136.7159	135.2512
R _p (%)	7.53	7.33	7.49	7.56	7.52
R _{wp} (%)	7.02	6.62	10.29	6.36	7.02
GoF (χ^2)	0.1916	0.227	0.9562	0.1725	0.1916



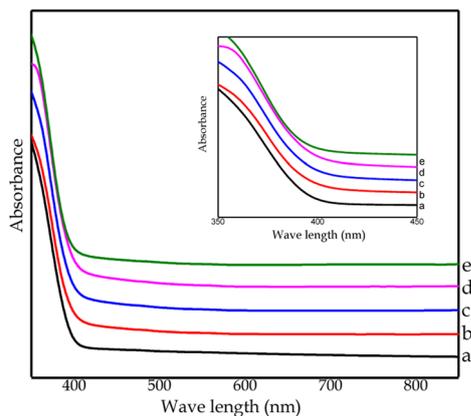


Fig. 7 DRS UV-Vis spectrum of all samples: (a) CTO, (b) NCT0.25, (c) NCT0.5, (d) NCT0.75, and (e) NCT1.

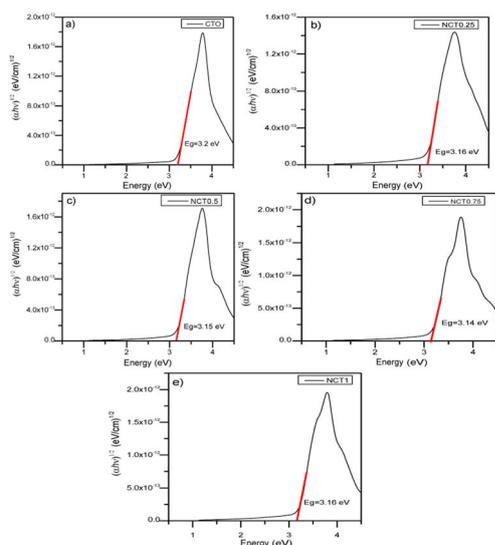


Fig. 8 Band gap energy of all samples: (a) CTO, (b) NCT0.25, (c) NCT0.5, (d) NCT0.75, and (e) NCT1.

absorption peaks in the range of wavenumbers between 400 and 800 cm^{-1} indicating strong strain vibrations of the Ti-O and Ti-O-Ti bonds in N-doped TiO_2 and TiO_2 .^{33,40} Wave-number 505.31–680.83 cm^{-1} is a characteristic of Ti-N bond vibrations.³²

Surface area analyzer (SSA) analysis

SAA characterization was used to measure pore size (cc/g), specific surface area ($\text{m}^2 \text{g}^{-1}$), and pore diameter (D) in N-doped TiO_2 samples with the addition of CTAB and N-doped TiO_2 without the addition of synthesized CTAB.⁴¹ The surface area and porosity of the N-doped TiO_2 sample with and without the addition of CTAB were investigated through the nitrogen adsorption-desorption isotherm shown in Fig. 10. The adsorption-desorption isotherms shown in Fig. 10a and b illustrate the type IV isotherm which is a characteristic of

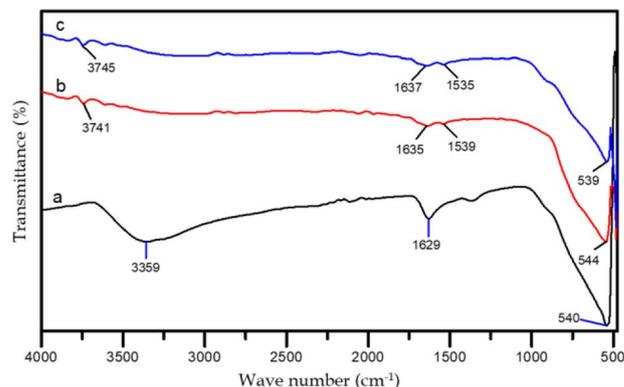


Fig. 9 FTIR spectrum of samples: (a) CTO, (b) NCT0.75, and (c) NTi0.75.

mesoporous materials. Based on the data in Table 4, it is shown that the specific surface area of the NCT0.75 sample ($95.02 \text{ m}^2 \text{ g}^{-1}$) is larger compared to that of the NTi0.75 sample, which is $90.97 \text{ m}^2 \text{ g}^{-1}$. The resulting pore sizes of the NCT0.75 and NTi0.75 samples were 8.021 nm and 5.161 nm, respectively. It is proven that the synthesized material had pores with mesoporous sizes because of having pores in the size range of 2 to 50 nm. The NCT0.75 sample had a higher specific surface area than the NTi0.75 sample. This proves that adding CTAB played a role as a pore-forming agent in the sample to increase the specific surface area of the sample and its photocatalytic activity.¹⁶

Porous materials with a large specific surface area have not maximized the photocatalytic activity of TiO_2 . Li, Xiao *et al.*¹⁹ synthesized mesoporous TiO_2 spheres by further modifying nitrogen doping with the addition of CTAB as a pore-forming agent, resulting in a porous material that has a high specific surface area, homogeneous, uniform, and high concentration to enhance TiO_2 photocatalytic activity in degrading pollutants. The relatively high specific surface area of the porous N-doped TiO_2 can facilitate the diffusion of reactant molecules and offer more active sites for adsorption so that the photocatalytic process is more efficient.

So that it can be confirmed that the photocatalytic activity can be increased due to 2 factors: firstly the addition of a pore-forming agent (CTAB) to produce many and uniform pores, thus increasing the specific surface area of the porous N-doped TiO_2 sample, and secondly doping using nitrogen to reduce the bandgap energy TiO_2 . Therefore, TiO_2 is able to optimize its

Table 3 Identification of FTIR vibration in TiO_2 (CTO), N-doped TiO_2 samples with the addition of CTAB (NCT0.75) and N-doped TiO_2 without CTAB (NTi0.75)

Wave number (cm^{-1})			Vibration type
CTO	NCT0.75	NTi0.75	
3359	3741	3745	Strain vibration of O-H
1629	1635	1637	Bending vibration of O-H
—	1539	1535	Vibration of N-H
540	544	539	Vibration of Ti-O, Ti-O-Ti, dan Ti-N



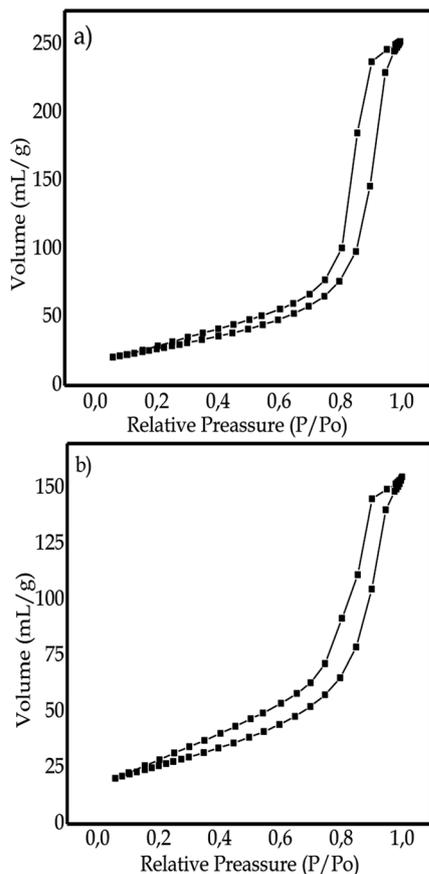


Fig. 10 BET isotherm curve of samples: (a) NCT0.75, (b) NTi0.75.

performance as a photocatalyst with a dual role of adsorption and photodegradation as well.

Based on Fig. 11, it can be explained that there was an interaction between TiO_2 and surfactants. The surfactant forms a micelle to reduce surface tension. The positively-charged TiO_2 material interacts electrostatically through O^+ ions on H_2O^+ bound to the surface of TiO_2 with Br^- ions which are the hydrophilic part of CTAB.⁴² In the hydrothermal process, a crystallization process undergoes structural rearrangement due to heating to form a crystal core. At this stage, crystal growth occurs. In the next stage, calcination takes place with the purpose of decomposing the surfactants remaining in the pore material. The calcination in this study occurred at the temperature of 500 °C for 1 h which was much higher than the CTAB thermal decomposition temperature (300 °C). The calcination is expected to decompose the CTAB surfactant into O_2 , CO_2 , and H_2O gases to leave open pores with a size matching the size of the surfactant.⁴³

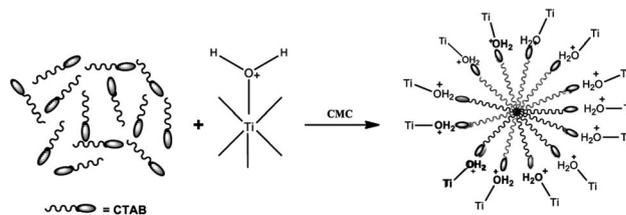


Fig. 11 Interaction between TiO_2 and CTAB on micelle formation.⁴²

Photocatalytic activity test

The synthesized N-doped TiO_2 sample with the addition of CTAB (NCT0.75) and control TiO_2 (CTO) were tested for a photocatalytic activity to reduce 10 mg L^{-1} of $\text{Cr}(\text{vi})$ metal ions irradiated under a 24 watt LED lamp with different irradiation times of 0 min and 240 min. The $\text{Cr}(\text{vi})$ solution was prepared under acidic conditions until the pH reached 2. This is because the lower the pH, the higher the concentration of H^+ ions. It is known from the overall $\text{Cr}(\text{vi})$ reduction reaction equation that protons (H^+) are needed in this process. Therefore, with an increase in the number/availability of protons, the equilibrium shifts to the right, and the rate of the $\text{Cr}(\text{vi})$ reduction reaction increases.⁴⁴

The photocatalytic test was initiated with adsorption-desorption equilibrium in the dark for 210 min to observe the adsorption/absorption effect of the catalyst on the $\text{Cr}(\text{vi})$ solution. The metal ion $\text{Cr}(\text{vi})$ has a specific complexing reagent used in quantitative analysis, namely 1,5-diphenylcarbazide. The Cr -diphenylcarbazide complex produces a purplish pink color, the complementary color of the color absorbed by the complex. The color of a complex is caused by the transition of electrons from the ground state to a higher energy level. When the electron transition requires radiation in the visible light range, the color of the complex changes. Transition in this complexing reaction occurs with a charge transfer between metal to ligand and ligand to metal. The reaction can be seen in Fig. 12. The absorbance of the complex solution was measured at a wavelength of 543 nm using a UV-Vis spectrophotometer.⁴⁵

The photocatalytic test results in Fig. 13 compare the adsorption and reduction percentages of $\text{Cr}(\text{vi})$ for each catalyst at the 0th and 240th minute of irradiation. The percentage reduction of $\text{Cr}(\text{vi})$ ions from CTO, NCT0.25, NCT0.5, NCT0.75, NCT1, and NTi0.75 after 240 minutes of irradiation were 36.89%, 48.68%, 80.22%, 89, 42%, 81.38%, and 72.66%. The adsorption ability of the samples CTO, NCT0.25, NCT0.5, NCT0.75, NCT1, and NTi0.75 was 3.19%, 24.91%, 27.86%, 30, 66%, 29.3%, and 20.47%, respectively, for 210 minutes in the dark.

Table 4 BET results for pore diameter, pore volume, and specific surface area

No.	Sample	Specific surface area ($\text{m}^2 \text{g}^{-1}$)	Pore diameter (nm)	Pore volume (mL g^{-1})
1	N-doped TiO_2 with CTAB (NCT0.75)	95.02	8.021	0.3902
2	N-doped TiO_2 without CTAB (NTi0.75)	90.97	5.161	0.2398



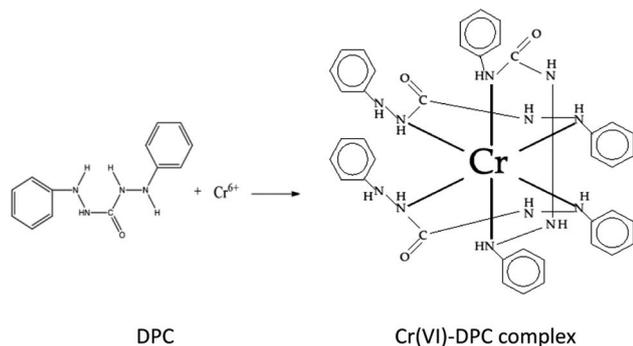


Fig. 12 Complexing reaction of 1,5-diphenyl carbazide with Cr(vi) ion.⁴⁶

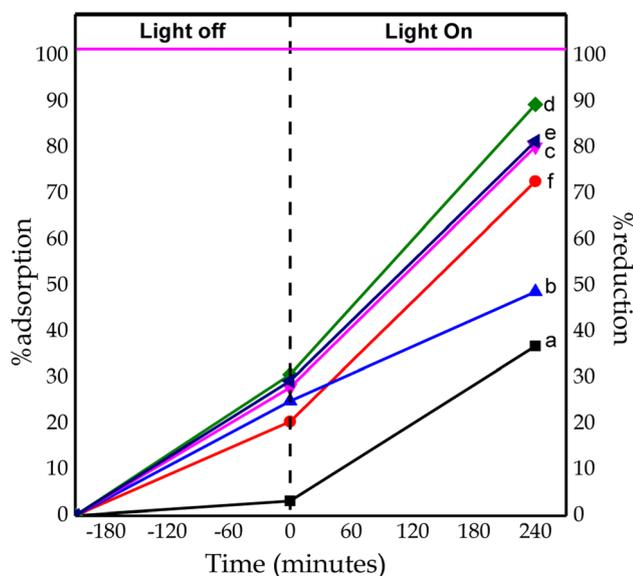


Fig. 13 Percentage of adsorption and reduction of Cr(vi) ions by catalysts: (a) CTO, (b) NCT0.25, (c) NCT0.5, (d) NCT0.75, (e) NCT1, and (f) NTi0.75.

It can be seen from Fig. 13 NCT0.75 had the most optimum percentage of adsorption and reduction due to the addition of nitrogen dopants with the optimum volume of 0.75 mL. Modification using nitrogen dopants can narrow the TiO₂ band gap. So, excited electrons from the valence band can be accepted by nitrogen above the valence band, which is then transferred to the conduction band with lower energy absorption. This can effectively inhibit the occurrence of photoexcited holes and also reduce electron-hole recombination. So it can increase the photocatalytic activity of TiO₂.⁴⁷ However, the nitrogen volume above 1 mL decreases the photocatalytic activity. This is because the higher volume of added dopant nitrogen beyond its optimum limit resulted in a partial covering of the TiO₂ surface where the Cr(vi) reduction reaction occurred, thereby increasing the chances of electron-hole recombination during the photocatalytic process.⁴⁴ As a result, the photocatalytic activity of Cr(vi) reduction decreased. In addition, the high percentage of photocatalytic activity in the NCT0.75 sample was caused by the

formed anatase/brookite TiO₂ heterostructured material, which was able to inhibit the rate of electron-hole recombination to extend the lifetime of a photocatalyst. This was associated with the synergistic effect of combining anatase and brookite TiO₂ semiconductors affecting the increase in photocatalyst activity.^{48,49}

In Fig. 14, it can be seen that electrons were able to move from the TiO₂ brookite conduction band to the conduction band of TiO₂ anatase. In contrast, h⁺ was able to move from the valence band of TiO₂ anatase to the valence band of TiO₂ brookite resulting in a high separation efficiency of electron-hole pairs. The anatase/brookite TiO₂ heterojunction material had a high interfacial area, so the efficiency of separating electron-hole pairs and the photocatalytic activity increases.⁵⁰

The increase in photocatalytic activity could also be influenced by adsorption factors, namely by adding CTAB as a pore-forming agent. The pores formed in the sample function as adsorbents that could reduce the concentration of the heavy metal Cr(vi). So the photocatalytic activity increased.⁸ It can be seen from the results of this study that the NCT0.75 sample had a higher reduction percentage (89.42%) compared to that of the NTi0.75 sample (72.66%). The difference between these two samples was due to adding CTAB to N-doped TiO₂. Research by Li Hongliang *et al.* (2016) stated that CTAB added-TiO₂ samples provided the highest photocatalytic percentage compared to that of other samples. This was attributed to the addition of CTAB as a pore-forming agent, which could increase the surface area of the sample.⁸

The percentage reduction increased with increasing irradiation time. By increasing the irradiation time, the catalyst receives light for a longer time, followed by generating more electrons. The resulting electrons play a role in the reduction reaction of Cr(vi). The overall reactions are as follows:

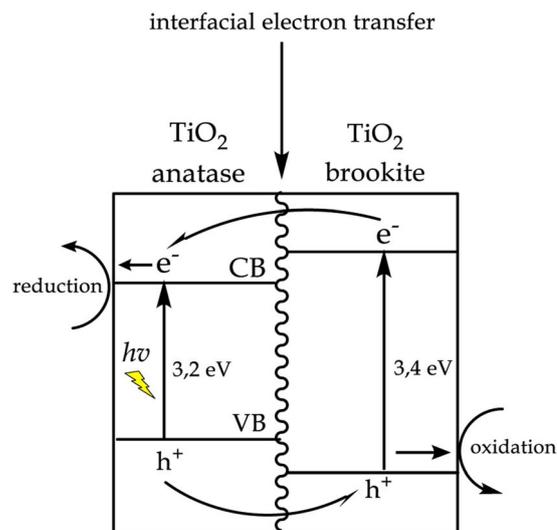
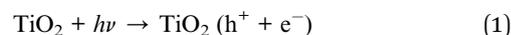
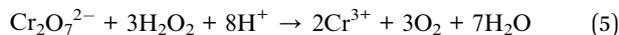
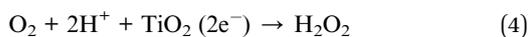
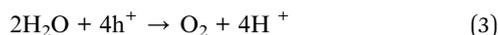
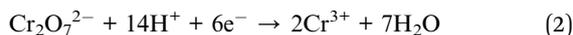


Fig. 14 Charge transfer scheme at anatase/brookite TiO₂ heterojunction.





Conclusions

According to the results, it can be concluded that the porous N-doped TiO₂ photocatalyst was successfully synthesized using the green synthesis-hydrothermal method. Adding CTAB as a pore-forming agent to the photocatalyst gave mesoporous structures with pore sizes ranging from 5–8 nm. It also increased the specific surface area of N-doped TiO₂, namely 90–95 m² g⁻¹. The morphology of the synthesized TiO₂ nanoparticles was spherical/spherical, with a size distribution of 10–30 nm. The NCT0.75 photocatalyst with the addition of 0.75 mL nitrogen dopants was the most optimum dopant because of having the best photocatalytic activity in reducing Cr(vi) heavy metal ions with a reduction percentage of 89.42% and an adsorption percentage of 30.66%. The porous N-doped TiO₂ photocatalyst had the optimal photocatalytic activity in the presence of CTAB. So it exhibited a dual role of adsorption and photoreduction of Cr(vi) heavy metal ions. The photocatalytic activity was increased by adding various nitrogen dopants in porous TiO₂. The percentage reduction of Cr(vi) ions from CTO, NCT0.25, NCT0.5, NCT0.75, NCT1, and NTi0.75 after 240 minutes of irradiation were 36.89%, 48.68%, 80.22%, 89, 42%, 81.38%, and 72.66%, respectively.

Author contributions

Conceptualization: D. V. W.; investigation: A. S.; methodology: D. V. W., Y. E. P.; supervision: D. V. W., Y. E. P.; visualization: A. S.; writing-original draft: D. V. W., A. S.; writing-review & editing: M.

Conflicts of interest

There are no conflicts to declare.

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