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# Magnetization reversal of perpendicular magnetic anisotropy regulated by ferroelectric polarization in CoFe<sub>3</sub>N/BaTiO<sub>3</sub> heterostructures: first-principles calculations

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Exploring the electric-field switching of perpendicular magnetic anisotropy (PMA) in multiferroic heterostructures has important physical significance, which attracts great interest due to its promising application for energy-efficient information storage. Herewith, we investigate the effect of ferroelectric polarization on magnetic anisotropy in CoFe<sub>3</sub>N/BaTiO<sub>3</sub> heterostructures using first-principles calculations. The calculations reveal that the magnetic anisotropy of CoFe<sub>3</sub>N can be regulated by ferroelectric polarization of BaTiO<sub>3</sub>. When the ferroelectric polarization reverses, the PMA of FeCo–TiO<sub>2</sub> and FeN–BaO configurations remains, but in the FeN–TiO<sub>2</sub> and FeCo–BaO cases, magnetic anisotropy inverses between out-of-plane and in-plane direction. Further orbital-resolved analysis indicates that the transition of magnetic anisotropy is mainly attributed to the orbital hybridization of interfacial Fe/Co atoms with O atoms induced by the magnetoelectric effect. This study may open an effective approach toward modulating PMA and lays a foundation to the development of low energy consumption memory devices.

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## Introduction

With the rapid development of information technology and the continuous advancement of spintronics, perpendicular magnetic anisotropy (PMA) has attracted intensive attention with potential application in energy-efficient information memory.<sup>1–3</sup> PMA is beneficial to obtain higher storage density, higher thermal stability and lower critical switching current density in non-volatile magnetic random access memory (MRAM).<sup>4–6</sup> PMA mainly arises at ferromagnetic/oxide interfaces, such as Fe/MgO,<sup>6,7</sup> CoFeB/MgO,<sup>5</sup> CoFeB/BaTiO<sub>3</sub>,<sup>8</sup> Mn<sub>3</sub>Ga/SrTiO<sub>3</sub>,<sup>9</sup> and (Co/Pt)<sub>3</sub>/PMN-PT.<sup>10</sup> It is also interesting that the magnetism of memory devices could be controlled by using an electric field through magnetoelectric (ME) coupling in multiferroic heterostructures. The ME effect at ferromagnetic/ferroelectric heterostructures also reduces the energy dissipation by Joule heating, which becomes another hot topic of information memory.<sup>11–15</sup> Various mechanisms on ME effect have been proposed and confirmed, such as strain,<sup>16,17</sup> charge,<sup>18</sup> and exchange-coupling.<sup>19–21</sup> More importantly, multiferroic heterostructures with PMA provide more effective method for achieving the magnetization switching *via* electric field, which will not only save energy consumption, but also expand the future application for non-volatile MRAM device.<sup>22,23</sup>

Among ferromagnets, anti-perovskite material Fe<sub>4</sub>N has been reported as a very attractive candidate for spintronics applications due to high spin polarization, small coercivity and high Curie temperature.<sup>24–26</sup> In our previous experiment, epitaxial Fe<sub>4</sub>N films can be grown on MgO, PMN-PT, and SrTiO<sub>3</sub> substrates and they all exhibit good interface and magnetic properties.<sup>27–29</sup> Experiment results show that the ME effect and magnetic anisotropy of Fe<sub>4</sub>N/PMN-PT heterostructure can be tailored by electric field,<sup>27,30</sup> further confirmed by theoretical calculation.<sup>31</sup> Fe<sub>4</sub>N/MgO/Fe<sub>4</sub>N magnetic tunnel junction (MTJ) has been predicted to produce an ultrahigh tunnel magnetoresistance (TMR),<sup>32</sup> Fe<sub>4</sub>N/Alq<sub>3</sub>/Co and LSMO/C<sub>60</sub>/Fe<sub>4</sub>N-MTJs exhibit a negative TMR and a regulable interface.<sup>33,34</sup> PMA has been obtained in Fe<sub>4</sub>N/BiFeO<sub>3</sub>, Fe<sub>4</sub>N/MgO, and Fe<sub>4</sub>N/PMN-PT heterostructure and it also can be tuned by strain, interfacial oxidation, ferroelectric polarization, and electric field.<sup>31,35–38</sup> However, the electric field switching of PMA does not achieve in Fe<sub>4</sub>N-based heterostructure by ferroelectric polarization. Bulk Fe<sub>4</sub>N is cubic symmetry and it does not have PMA, but the substitution of Co in the Fe<sub>4</sub>N lattice induces a large PMA of tetragonal CoFe<sub>3</sub>N.<sup>39</sup> By adsorbing organic molecules, organic/CoFe<sub>3</sub>N spinterface exhibits enhanced PMA.<sup>40</sup> Moreover, the Mn-substitution doping at the interfacial Fe<sub>II</sub> position in the Fe<sub>4</sub>N/BaTiO<sub>3</sub> heterostructure may obtain a large ME effect.<sup>41</sup> In order to further improve the interfacial PMA and achieving the electric field switching of PMA, the substitute CoFe<sub>3</sub>N is introduced as the ferromagnetic layer of multiferroic

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heterostructure, which has a great application prospect for spintronics.

In this work, we construct the  $\text{CoFe}_3\text{N}/\text{BaTiO}_3$  multiferroic heterostructures and investigate the impact of ferroelectric polarization on magnetic anisotropy using first-principles calculations. Our studies indicate that the magnetic anisotropy of  $\text{CoFe}_3\text{N}$  can be regulated by ferroelectric polarization of  $\text{BaTiO}_3$ . Ferroelectric polarization reversal makes flipping of magnetization between the out-of-plane and in-plane direction in  $\text{FeN-TiO}_2$  and  $\text{FeCo-BaO}$  interface configurations. Further orbital-resolved analysis indicates that the transition of magnetic anisotropy is mainly due to the strong interfacial ME coupling and orbital hybridization between interfacial Fe/Co atom and O atom. These results lay the foundation for realizing electric-field control of PMA in multiferroic heterostructures.

## Computational details and models

First-principles calculations are carried out based on density functional (DFT) theory using Vienna *Ab initio* Simulation Package code.<sup>42–44</sup> Exchange and correlation effects are accounted for by the generalized gradient approximation (GGA) as parameterized by Perdew, Burke, and Ernzerhof (PBE).<sup>45</sup> Although the PBE functional has an overestimation for unit cell volume,<sup>46</sup> extensive studies demonstrate that it is suitable for the study of multiferroic heterostructure.<sup>17,47,48</sup> Structural relaxations are performed until the force becomes less than  $10^{-2}$  eV  $\text{\AA}^{-1}$  and the change in the total energy between two ionic relaxation steps is smaller than  $10^{-5}$  eV. An energy cutoff of 500 eV and  $9 \times 9 \times 9$ ,  $11 \times 11 \times 1$   $k$ -point meshes are used for the bulk and heterostructures, respectively. The  $\text{CoFe}_3\text{N}/\text{BaTiO}_3$  heterostructures comprise seven-layered  $\text{CoFe}_3\text{N}$  and seven-layered  $\text{BaTiO}_3$ , followed by a 15  $\text{\AA}$  vacuum layer. The experimental lattice constant of bulk  $\text{BaTiO}_3$  is 3.99  $\text{\AA}$  and the in-plane lattice constant of bulk  $\text{CoFe}_3\text{N}$  is 3.78  $\text{\AA}$ ,<sup>39</sup> which has a lattice mismatch of 5.3%. Here, we build the  $\text{CoFe}_3\text{N}/\text{BaTiO}_3$  heterostructures is based on the previous experiment research of  $\text{Fe}_3\text{N}/\text{oxide}$  heterostructure.<sup>27,28,30</sup> The binding energy ( $E_B$ ) calculation is performed to display the structural stability of different interface configurations. The magnetic anisotropy energy (MAE) is determined using the magnetic force theorem with spin-orbit coupling (SOC).<sup>49–51</sup> The MAE depends on the nonzero coupling matrix element between the occupied and unoccupied d-orbital states.<sup>51</sup> The total MAE values of  $\text{CoFe}_3\text{N}/\text{BaTiO}_3$  heterostructures are calculated from the energy difference between the magnetic moment aligning in the in-plane ( $x$  axis) and out-of-plane ( $z$  axis) orientations. The positive value of MAE represents PMA and negative value represents in-plane magnetic anisotropy (IMA).

## Results and discussion

Prior to investigating the regulated magnetic anisotropy by ferroelectric polarization, we first discuss the properties of bulk  $\text{CoFe}_3\text{N}$  and  $\text{BaTiO}_3$ , as shown in Fig. 1. In tetragonal  $\text{CoFe}_3\text{N}$ , the corner site is occupied by Fe atom, named it  $\text{Fe}_I$ , the Fe

atoms at face-centered site are named by  $\text{Fe}_{II}$ , and another face-centered site is occupied by Co atom, as shown in Fig. 1(a). The strong hybridization between  $\text{Fe}_{II}$  and N atoms makes distinct density of states in  $\text{Fe}_I$  and  $\text{Fe}_{II}$  atoms,<sup>32,52</sup> so  $\text{Fe}_I$  and  $\text{Fe}_{II}$  atoms exhibit outstanding difference in magnetic properties. In bulk  $\text{CoFe}_3\text{N}$ , the magnetic moments of  $\text{Fe}_I$ ,  $\text{Fe}_{II}$ , and Co atoms are 2.987  $\mu_B$ , 2.229  $\mu_B$ , and 1.180  $\mu_B$ , respectively. In Fig. 1(c), the MAE results of  $\text{CoFe}_3\text{N}$  show that  $\text{Fe}_I$ ,  $\text{Fe}_{II}$ , and Co atoms have a noticeable difference.  $\text{Fe}_I$  atom has an IMA contribution whereas  $\text{Fe}_{II}$  atom exhibits a PMA. Furthermore, Co atom shows a large PMA of 0.319 meV. We then display the orbital-resolved MAE of Fe and Co atoms in Fig. 1(d)–(f) to analysis the orbital contribution for MAE. In  $\text{Fe}_I$ , nonzero coupling matrix element  $\langle x^2 - y^2 | \hat{L}_z | xy \rangle$  favors IMA, but it turns to support PMA in  $\text{Fe}_{II}$  and Co atoms. The difference in chemical environment of  $\text{Fe}_I$ ,  $\text{Fe}_{II}$ , and Co atoms determines the d-orbital distribution near the Fermi level,<sup>32,38,52</sup> finally altering the magnetic moment and MAE of Fe and Co atoms. Fig. 1(g) displays the total density of states (DOS) and partial DOS of bulk  $\text{CoFe}_3\text{N}$ . The results show that there is spin majority states of  $\text{Fe}_{II}$  and Co atoms pass through the Fermi level whereas there are not spin majority d states near the Fermi level in  $\text{Fe}_I$  atom, which leads to the distinct magnetic properties. Fig. 1(b) shows that bulk  $\text{BaTiO}_3$  has a typical perovskite structure and the DOS result in Fig. 1(h) indicates that it is a non-magnetic insulator.<sup>53</sup>

We then built  $\text{CoFe}_3\text{N}/\text{BaTiO}_3$  heterostructures with ideal interface, and complex interface situations such as oxygen diffusion, disorder, and lattice distortion are not considered in this work. Four possible ideal interface structures are taken into account, namely,  $\text{FeCo-TiO}_2$ ,  $\text{FeN-TiO}_2$ ,  $\text{FeCo-BaO}$ , and  $\text{FeN-BaO}$  type interface, as shown in Fig. 2. The grey arrows indicate the polarization direction of  $\text{BaTiO}_3$ . The positive ferroelectric polarization along the  $z$  axis is pointing away from the  $\text{BaTiO}_3$ . It should be pointed that interfacial Fe atom in  $\text{FeCo}$  termination is  $\text{Fe}_I$  and interfacial Fe atom in  $\text{FeN}$  termination is  $\text{Fe}_{II}$ . In order to explore the influence of interface structure on magnetic anisotropy, we first analyze the interface bonding and structure difference. In  $\text{FeCo-TiO}_2$  configuration, interfacial Co atom occupies atop sites on Ti atom. Polarization reversal makes the interfacial Co atom shift away from the interface. In  $\text{FeN-TiO}_2$  configuration, interfacial  $\text{Fe}_{II}$  atom bonds with O atom. Such a Fe–O bond leads to d-orbital reconstruction, which in turn affects the magnetic properties of  $\text{CoFe}_3\text{N}$  part with the presence of ferroelectric polarization. Intriguingly, interfacial Co and O atoms are bonded strongly and the bond length is only 1.811  $\text{\AA}$  in  $\text{FeCo-BaO}$  ( $P_{\perp}$ ) case, making the overlap between Co-3d and O-2p states stronger, whereas Co and O atoms are far away from each other in  $\text{FeCo-BaO}$  ( $P_{\parallel}$ ) case. In  $\text{FeN-BaO}$  configuration, regardless of polarization direction, they all exhibit a large interface distance. The significant change of interface structure with the process of polarization reversal may cause the change in electronic structure, spin magnetism, and MAE.<sup>53,54</sup>

In Table 1, we first list the  $E_B$  of  $\text{CoFe}_3\text{N}/\text{BaTiO}_3$  heterostructures with different configurations to determine the stability of heterostructures, which is defined as  $E_B = E_{\text{CoFe}_3\text{N}/\text{BaTiO}_3} - E_{\text{CoFe}_3\text{N}} - E_{\text{BaTiO}_3}$ .  $E_{\text{CoFe}_3\text{N}}$  and  $E_{\text{BaTiO}_3}$  represent the



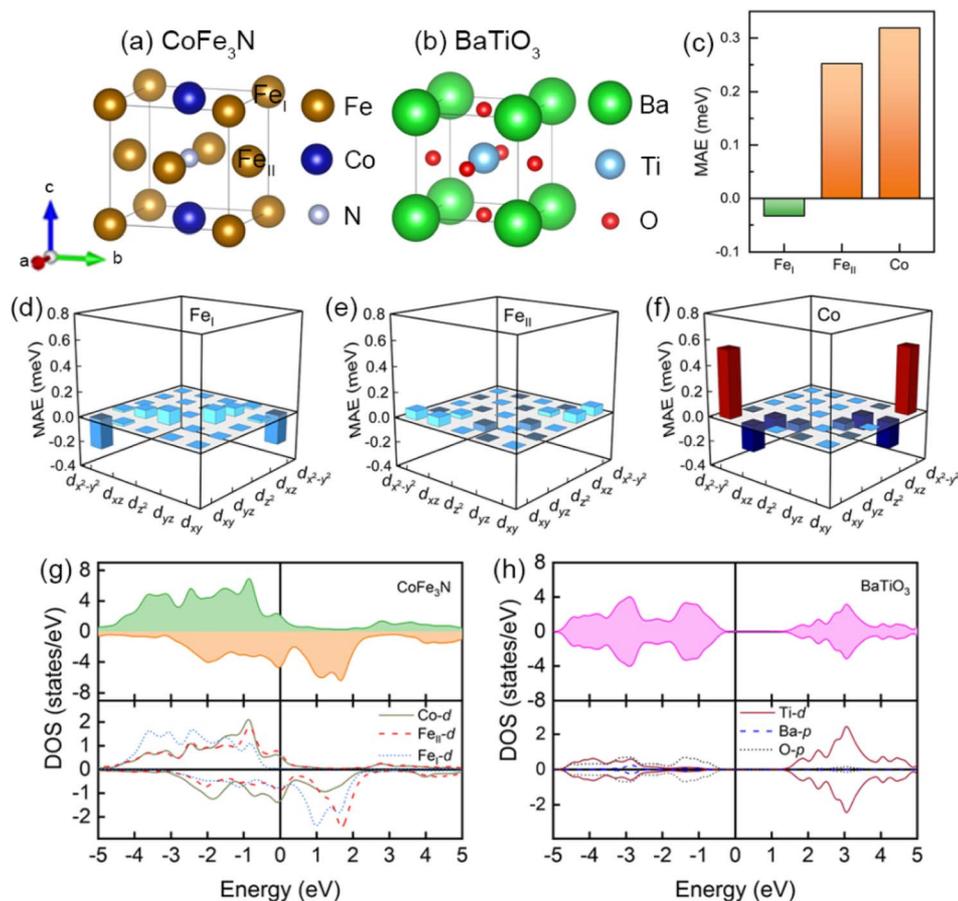


Fig. 1 (a) and (b) The lattice structure of bulk  $\text{CoFe}_3\text{N}$  and  $\text{BaTiO}_3$ . (c) MAE of Fe and Co atoms in bulk  $\text{CoFe}_3\text{N}$ . (d)–(f) Orbital-resolved MAE of Fe and Co atoms in bulk  $\text{CoFe}_3\text{N}$ . (g) and (h) Total and partial density of states (DOS) of bulk  $\text{CoFe}_3\text{N}$  and  $\text{BaTiO}_3$ .

energies of the surface of  $\text{CoFe}_3\text{N}$  part and  $\text{BaTiO}_3$  part, and  $E_{\text{CoFe}_3\text{N}/\text{BaTiO}_3}$  is the total energy of  $\text{CoFe}_3\text{N}/\text{BaTiO}_3$  heterostructure.<sup>47,53</sup> The negative  $E_B$  indicates that  $\text{FeCo-BaO}$  and  $\text{FeN-TiO}_2$  configurations are experimentally feasible.  $\text{FeCo-}$

$\text{TiO}_2$  ( $P_1$ ) and  $\text{FeN-BaO}$  ( $P_1$ ) configurations have positive values, indicating the process of polarization reversal may affect the interface stability.

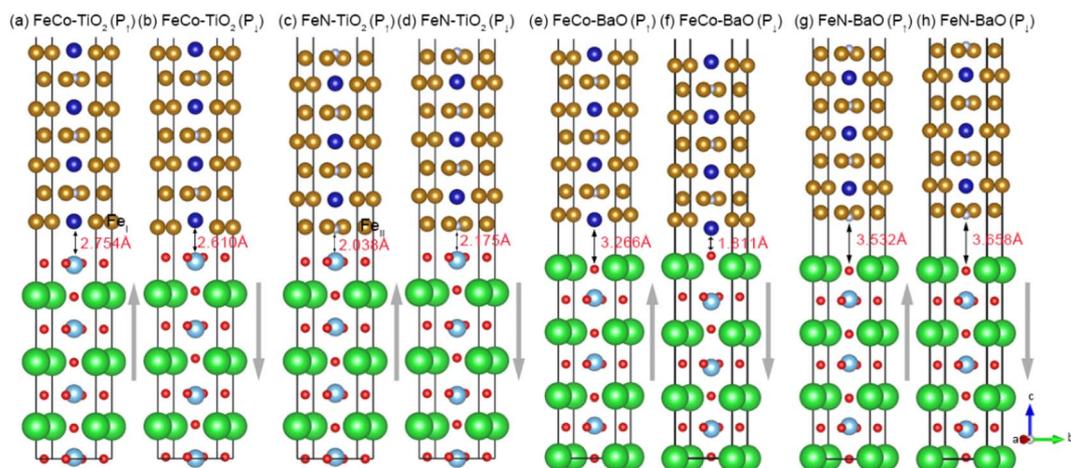


Fig. 2 The  $\text{CoFe}_3\text{N}/\text{BaTiO}_3$  heterostructures with different configurations. (a) and (b)  $\text{FeCo-TiO}_2$ , (c) and (d)  $\text{FeN-TiO}_2$ , (e) and (f)  $\text{FeCo-BaO}$ , and (g) and (h)  $\text{FeN-BaO}$ . The grey arrows indicate the polarization direction of  $\text{BaTiO}_3$ . The interfacial Fe and Co atoms and interface distance are marked.



**Table 1**  $E_B$  (eV), magnetic moment ( $\mu_B$ ), and MAE ( $\text{mJ m}^{-2}$ ) of  $\text{CoFe}_3\text{N}/\text{BaTiO}_3$  heterostructures with different configurations

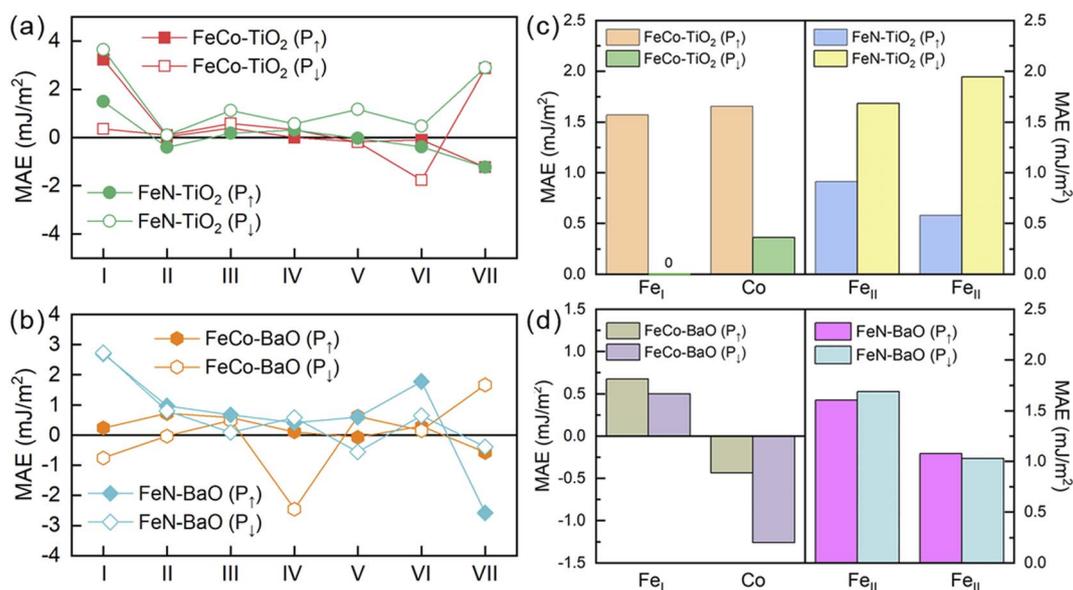
Configuration	$E_B$	$\mu_s^{\text{Ti/Ba}}$	$\mu_s^{\text{O}}$	$\mu_s^{\text{Fe}_I/\text{Fe}_{II}}$	$\mu_s^{\text{Co}}$	MAE
$\text{FeCo-TiO}_2$ ( $P_\uparrow$ )	0.181	-0.229	0.012	3.114	1.206	0.978
$\text{FeCo-TiO}_2$ ( $P_\downarrow$ )	-0.062	-0.116	0.012	3.251	-0.562	1.138
$\text{FeN-TiO}_2$ ( $P_\uparrow$ )	-2.476	-0.025	0.056	2.610	—	-0.043
$\text{FeN-TiO}_2$ ( $P_\downarrow$ )	-1.302	0.006	0.049	2.646	—	5.139
$\text{FeCo-BaO}$ ( $P_\uparrow$ )	-0.338	-0.006	-0.158	2.956	1.257	0.894
$\text{FeCo-BaO}$ ( $P_\downarrow$ )	-1.336	0.016	0.158	2.846	1.591	-0.181
$\text{FeN-BaO}$ ( $P_\uparrow$ )	-0.367	-0.001	-0.138	2.476	—	2.532
$\text{FeN-BaO}$ ( $P_\downarrow$ )	1.005	-0.000	0.004	2.467	—	1.958

Table 1 also gives the magnetic moments of the interfacial Ti, Ba, O, Fe, and Co atoms in  $\text{CoFe}_3\text{N}/\text{BaTiO}_3$  heterostructures with  $P_\uparrow$  and  $P_\downarrow$ . The appearance of magnetic moment in Ti and O atoms can be attributed to the orbital hybridization with Fe/Co atom, which demonstrates strong ME effect at the interface.<sup>48,55</sup> In  $\text{FeCo-TiO}_2$  configuration, the magnetic moment of  $\text{Fe}_I$  atom has a marked increase compared to the bulk  $\text{CoFe}_3\text{N}$ , owing to the strong Fe 3d and O 2p orbital hybridization. The polarization reversal gives rise to an inversion of Co magnetic moment from  $1.206 \mu_B$  ( $P_\uparrow$ ) to  $-0.562 \mu_B$  ( $P_\downarrow$ ), indicating that the magnetic structure can be regulated by  $\text{BaTiO}_3$  polarization. In  $\text{FeN-TiO}_2$  configuration, the magnetic moment of  $\text{Fe}_{II}$  atom has a tiny change with the polarization reversal. In  $\text{FeCo-BaO}$  configuration, the magnetic moment of O atom can be switched from parallel to antiparallel to the Co atom. This means that a strong ME coupling is formed in  $\text{FeCo-BaO}$  configuration, which has a good application prospect in spintronics. In addition, the total MAEs of  $\text{CoFe}_3\text{N}/\text{BaTiO}_3$  heterostructures are also shown in Table 1. For  $\text{FeN-TiO}_2$  and  $\text{FeCo-BaO}$  configuration, by changing the polarization direction, the easy magnetic axis is switched between out-of-plane direction and in-plane direction.

Especially, in  $\text{FeN-TiO}_2$  configuration, negative polarization makes the PMA increase significantly, confirming that polarization reversal produces a sizable change in magnetic anisotropy. However, regardless of polarization direction, PMA remains in  $\text{FeCo-TiO}_2$  and  $\text{FeN-BaO}$  configurations.

To further clarify the origin of MAE, layer-resolved MAEs are shown in Fig. 3(a) and (b). For both polarization directions, MAE changes mainly in the interfacial and surface layer, owing to the anisotropic nature resides at the interface or surface.<sup>48</sup> In Fig. 3(a), interfacial MAE has a sharp decline in  $\text{FeCo-TiO}_2$  configuration when the magnetoelectric effect is applied as  $P_\uparrow \rightarrow P_\downarrow$ . In  $\text{FeN-TiO}_2$  configuration, the magnitude of MAE in every layer has an increase with the polarization reversal. In Fig. 3(b),  $\text{FeCo-BaO}$  configurations with  $P_\uparrow$  and  $P_\downarrow$  states exhibit distinct trends in interfacial MAE, being positive and negative in sign, respectively. This indicates that the magnetization direction can reorient from out-of-plane easy axis to in-plane by reversing the polarization direction. In  $\text{FeN-BaO}$  configuration, interfacial MAE is almost unchanged as  $P_\uparrow \rightarrow P_\downarrow$ . Further, to get more insights, we analyze the atom-resolved MAE of interfacial Fe and Co atoms, as shown in Fig. 3(c) and (d). The interfacial Fe/Co atom plays an important role in the electronic structure near the Fermi level and interfacial magnetic anisotropy. In  $\text{FeCo-TiO}_2$  configuration, the MAE of  $\text{Fe}_I$  and Co atoms decrease severely. In  $\text{FeN-TiO}_2$  configuration, both  $\text{Fe}_{II}$  atoms favor the PMA and PMA increases at  $P_\downarrow$  state. In  $\text{FeCo-BaO}$  configuration,  $\text{Fe}_I$  favors PMA and yet Co supports IMA, which is contrary to the bulk  $\text{CoFe}_3\text{N}$ . When the polarization reverses, the opposite MAE of  $\text{Fe}_I$  and Co atoms brings about the changed sign of interfacial MAE. In  $\text{FeN-BaO}$  configuration, the MAE of interfacial  $\text{Fe}_{II}$  atoms changes little during  $P_\uparrow \rightarrow P_\downarrow$ .

To elucidate the effect of interfacial structure and polarization reversal on PMA, we show the d-orbital-resolved MAE of interfacial Fe and Co atoms in Fig. 4. According to the second-



**Fig. 3** (a) and (b) The layer-resolved MAE of  $\text{CoFe}_3\text{N}/\text{BaTiO}_3$  heterostructures with  $\text{TiO}_2$  and  $\text{BaO}$  terminations. (c) and (d) The MAE of interfacial Fe and Co atoms of  $\text{CoFe}_3\text{N}/\text{BaTiO}_3$  heterostructures with  $\text{TiO}_2$  and  $\text{BaO}$  terminations.



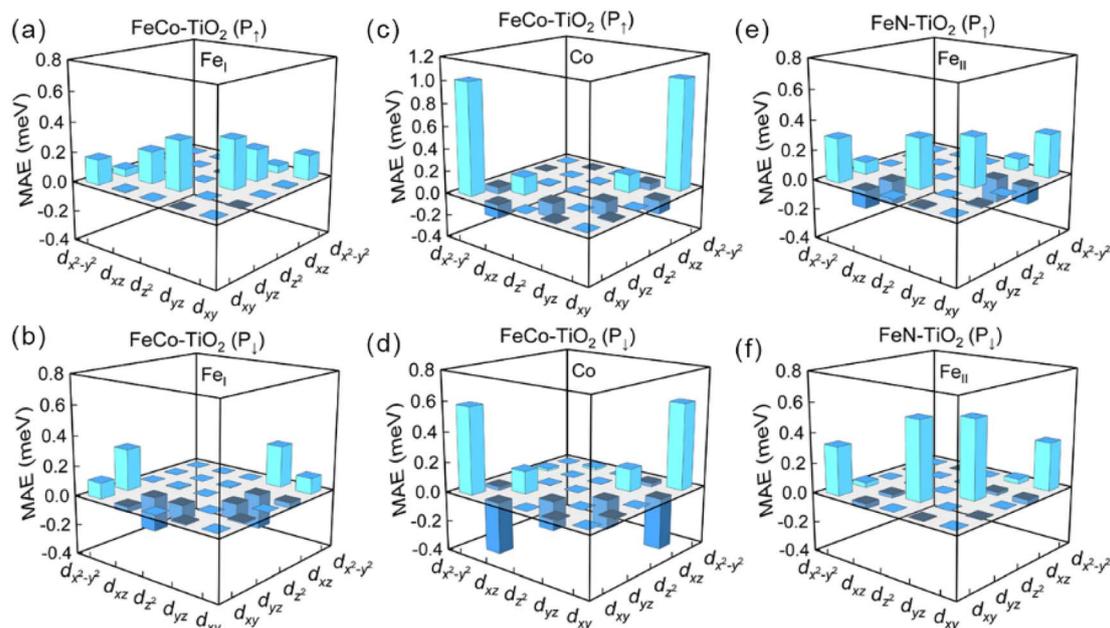


Fig. 4 The orbital-resolved MAE of interfacial Fe and Co atoms in  $\text{CoFe}_3\text{N}/\text{BaTiO}_3$  heterostructures with  $\text{TiO}_2$  termination. (a) and (b)  $\text{Fe}_I$  of  $\text{FeCo-TiO}_2$ , (c) and (d) Co of  $\text{FeCo-TiO}_2$ , (e) and (f)  $\text{Fe}_{II}$  of  $\text{FeN-TiO}_2$ .

order perturbation theory,<sup>50,51</sup> the MAE depends on the nonzero coupling matrix element between the occupied and unoccupied d-orbital states. The MAE can be defined as

$$\text{MAE} \propto \xi^2 \sum_{o,u} \frac{|\langle \psi_o | \hat{L}_z | \psi_u \rangle|^2 - |\langle \psi_o | \hat{L}_x | \psi_u \rangle|^2}{E_u - E_o}$$

where  $\psi_o$  and  $\psi_u$  indicate the occupied and unoccupied states with the energies  $E_o$  and  $E_u$ , respectively.  $\xi$  is the SOC constant.  $\hat{L}_{z(x)}$  is the  $z(x)$  component of the orbital angular momentum operator. These

nonzero coupling matrix elements include  $\langle xz | \hat{L}_z | yz \rangle = 1$ ,  $\langle x^2 - y^2 | \hat{L}_z | xy \rangle = 2$ ,  $\langle z^2 | \hat{L}_x | yz \rangle = 3$ ,  $\langle xy | \hat{L}_x | xz \rangle = 1$ , and  $\langle x^2 - y^2 | \hat{L}_x | yz \rangle = 1$ .<sup>50</sup>

Meanwhile, in order to better explain the PMA contribution of matrix elements, we analyze the DOS of interfacial Fe and Co atoms from the d-orbital perspective in Fig. 6. In  $\text{FeCo-TiO}_2$  ( $P_1$ ) case, for  $\text{Fe}_I$  atom, all nonzero matrix elements provide the positive contribution to MAE. Among them, the contribution of

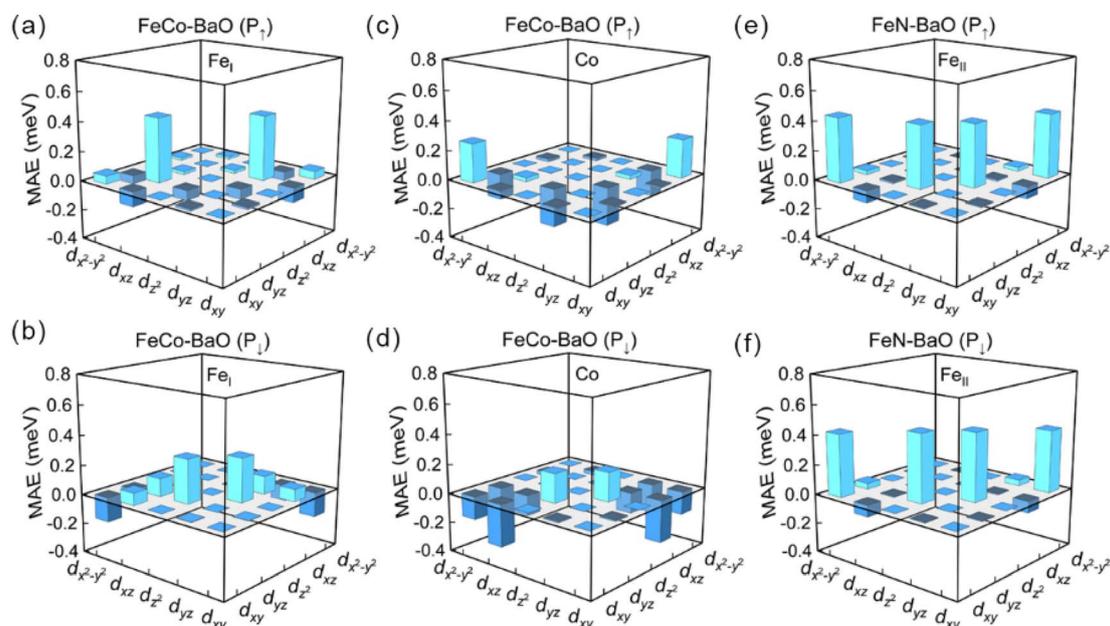


Fig. 5 The orbital-resolved MAE of interfacial Fe and Co atoms in  $\text{CoFe}_3\text{N}/\text{BaTiO}_3$  heterostructures with BaO termination. (a) and (b)  $\text{Fe}_I$  of  $\text{FeCo-BaO}$ , (c) and (d) Co of  $\text{FeCo-BaO}$ , (e) and (f)  $\text{Fe}_{II}$  of  $\text{FeN-BaO}$ .



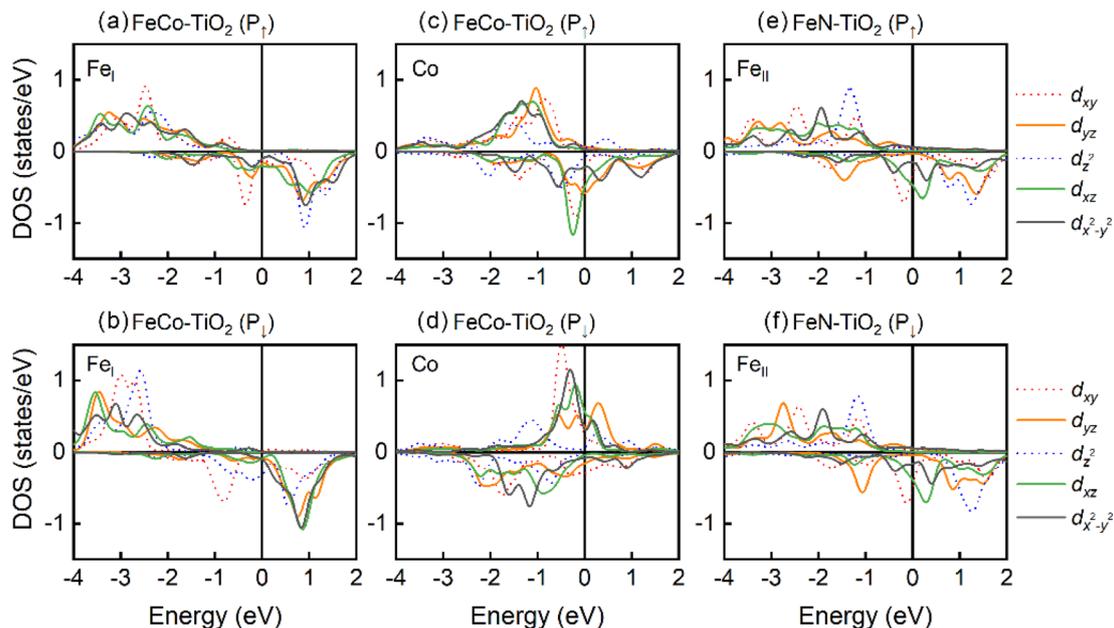


Fig. 6 The DOS of interfacial Fe and Co atoms in  $\text{CoFe}_3\text{N}/\text{BaTiO}_3$  heterostructures with  $\text{TiO}_2$  termination. (a) and (b)  $\text{Fe}_I$  of  $\text{FeCo-TiO}_2$ , (c) and (d) Co of  $\text{FeCo-TiO}_2$ , (e) and (f)  $\text{Fe}_{II}$  of  $\text{FeN-TiO}_2$ .

$\langle z^2|\hat{L}_x|yz\rangle$  is the most and it favors PMA. In  $\text{FeCo-TiO}_2$  ( $P_\downarrow$ ) case, these two terms of  $\langle z^2|\hat{L}_x|yz\rangle$  and  $\langle xz|\hat{L}_z|yz\rangle$  turn to support IMA, which cancels out the PMA contribution and makes the MAE of  $\text{Fe}_I$  atom zero. According to DOS of Fig. 6(a) and (b), during the  $P_\uparrow \rightarrow P_\downarrow$ , spin majority electron in  $d_{yz}$  occupied states of  $\text{Fe}_I$  atom disappears, which produces the negative contribution of  $\langle z^2|\hat{L}_x|yz\rangle$  and  $\langle xz|\hat{L}_z|yz\rangle$  terms. For Co atom, the  $\langle x^2-y^2|\hat{L}_z|xy\rangle$  matrix element always has a large PMA. However, the IMA

contribution of  $\langle xy|\hat{L}_x|xz\rangle$  increases as  $P_\uparrow \rightarrow P_\downarrow$ , causing the PMA of Co atom decreases, which is attributed to spin transition of  $d_{xz}$  occupied state near the Fermi level. In  $\text{FeN-TiO}_2$  configuration, polarization reversal ( $P_\uparrow \rightarrow P_\downarrow$ ) gives rise to the increase of PMA in interfacial  $\text{Fe}_{II}$  atom, which mainly comes from the growing PMA contribution of the matrix element  $\langle z^2|\hat{L}_x|yz\rangle$  and the reduced IMA contribution of  $\langle xz|\hat{L}_z|yz\rangle$  term. Based on the DOS of  $\text{Fe}_{II}$  atom,  $d_{yz}$  occupied state with spin

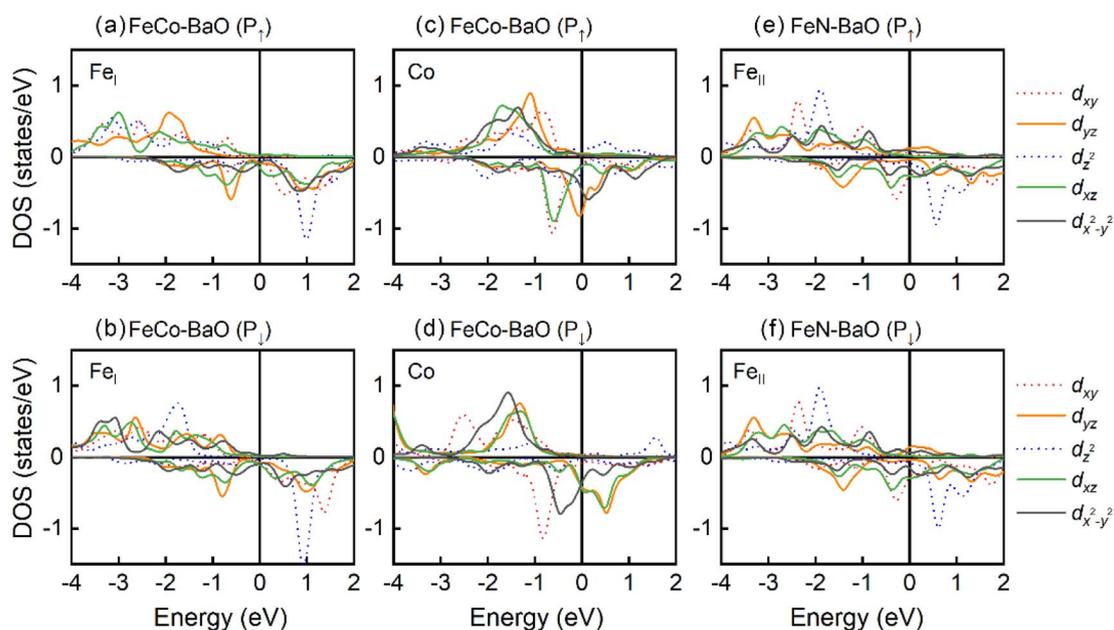


Fig. 7 The DOS of interfacial Fe and Co atoms in  $\text{CoFe}_3\text{N}/\text{BaTiO}_3$  heterostructures with BaO termination. (a) and (b)  $\text{Fe}_I$  of  $\text{FeCo-BaO}$ , (c) and (d) Co of  $\text{FeCo-BaO}$ , (e) and (f)  $\text{Fe}_{II}$  of  $\text{FeN-BaO}$ .



down shifts to the Fermi level, decreasing the energy level difference and leading to the growing PMA contribution of the matrix element  $\langle xz|\hat{L}_x|yz\rangle$  and the reduce of IMA contribution of  $\langle xz|\hat{L}_x|yz\rangle$ . It suggests that the MAE of the Fe<sub>II</sub> atom is more sensitive to the external condition, which is consistent with Fe<sub>4</sub>N/MgO heterostructure.<sup>38</sup>

Fig. 5 and 7 give the d-orbital-resolved MAE and DOS of interfacial Fe and Co atoms in FeCo–BaO and FeN–BaO structures. In FeCo–BaO configuration, during the P<sub>↑</sub> → P<sub>↓</sub>, most of zero matrix elements of Fe<sub>I</sub> atom changes the sign, which is ascribed to the appearance of majority 3d electrons including d<sub>yz</sub>, d<sub>z<sup>2</sup></sub>, d<sub>x<sup>2</sup>-y<sup>2</sup></sub> occupied states. Distinguishingly, the PMA contribution of  $\langle x^2-y^2|\hat{L}_z|xy\rangle$  of Co atom is no longer the dominant, even the contribution of  $\langle x^2-y^2|\hat{L}_z|xy\rangle$  turn into IMA with the polarization reversal. The inversion of MAE contribution of  $\langle x^2-y^2|\hat{L}_z|xy\rangle$  comes from the disappearance of d<sub>xy</sub> majority occupied states near the Fermi level. What's more, the MAE contributions of  $\langle x^2-y^2|\hat{L}_x|yz\rangle$  and  $\langle z^2|\hat{L}_x|yz\rangle$  terms are also changed in sign. On the basis of DOS result, the disappearance of d<sub>yz</sub> minority occupied states plays a key role in the change of the MAE sign. In FeN–BaO termination,  $\langle x^2-y^2|\hat{L}_z|xy\rangle$  and  $\langle z^2|\hat{L}_x|yz\rangle$  of interfacial Fe<sub>II</sub> atom have a positive contribution. The contribution of all matrix elements remains stable during the polarization reversal as a result of the weak ME coupling and unchanged DOS.

## Conclusions

In summary, the effect of ferroelectric polarization on magnetic anisotropy in CoFe<sub>3</sub>N/BaTiO<sub>3</sub> heterostructure has been demonstrated by first-principles calculations. We found that ferroelectric polarization reverses magnetic anisotropy of FeN–TiO<sub>2</sub> and FeCo–BaO configurations. The interfacial Fe<sub>I</sub>, Fe<sub>II</sub>, and Co atoms play a crucial role in determining the magnetic anisotropy of CoFe<sub>3</sub>N/BaTiO<sub>3</sub> heterostructures, which is attributed to the orbital hybridization between interfacial Fe/Co atom and O atom due to the ME coupling effect. This work paves the way for nonvolatile electrical control of multiferroic heterostructure and applications in spintronics devices.

## Author contributions

Zirun Li conducted first-principles calculations and drafted this paper. Bo Chen, Shimin Shan and Yongmei Zhang have taken part in the revision.

## Conflicts of interest

There are no conflicts of interest to declare.

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