



Cite this: *RSC Adv.*, 2023, **13**, 16773

## Self-healing hydrogels for bone defect repair

Weiwei Li,<sup>a</sup> Yanting Wu,<sup>a</sup> Xu Zhang,<sup>a</sup> Tingkui Wu,<sup>b</sup> Kangkang Huang,<sup>b</sup> Beiyu Wang<sup>\*b</sup> and Jinfeng Liao<sup>id \*a</sup>

Severe bone defects can be caused by various factors, such as tumor resection, severe trauma, and infection. However, bone regeneration capacity is limited up to a critical-size defect, and further intervention is required. Currently, the most common clinical method to repair bone defects is bone grafting, where autografts are the “gold standard.” However, the disadvantages of autografts, including inflammation, secondary trauma and chronic disease, limit their application. Bone tissue engineering (BTE) is an attractive strategy for repairing bone defects and has been widely researched. In particular, hydrogels with a three-dimensional network can be used as scaffolds for BTE owing to their hydrophilicity, biocompatibility, and large porosity. Self-healing hydrogels respond rapidly, autonomously, and repeatedly to induced damage and can maintain their original properties (*i.e.*, mechanical properties, fluidity, and biocompatibility) following self-healing. This review focuses on self-healing hydrogels and their applications in bone defect repair. Moreover, we discussed the recent progress in this research field. Despite the significant existing research achievements, there are still challenges that need to be addressed to promote clinical research of self-healing hydrogels in bone defect repair and increase the market penetration.

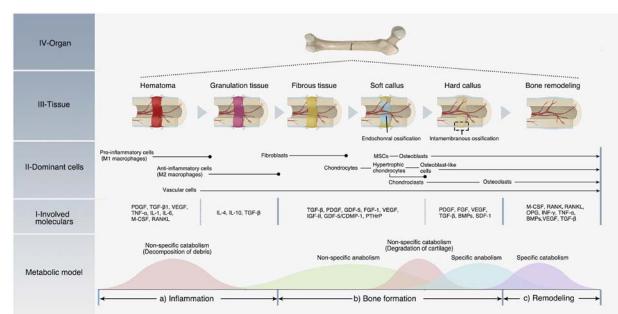
Received 15th March 2023  
 Accepted 24th May 2023

DOI: 10.1039/d3ra01700a  
[rsc.li/rsc-advances](http://rsc.li/rsc-advances)

## Introduction

Bone tissue is a hard connective tissue of the human body and is constantly reshaped throughout the life of an individual.<sup>1,2</sup> Bone tissue exerts both supporting and protective effects and can protect fragile organs within the body.<sup>3</sup> When bone defect areas are small, most bones can undergo self-healing without any treatment.<sup>4,5</sup> However, when bone tissue suffers damage beyond its ability to repair itself, bone damage can occur. Many factors affect the ability of bone tissue to regenerate and repair itself.<sup>6</sup> Severe bone defects can be caused by tumor resection,<sup>7,8</sup> severe trauma,<sup>9</sup> infection,<sup>10</sup> congenital malformation, osteogenesis imperfecta,<sup>11</sup> rheumatoid arthritis,<sup>12</sup> and osteoporosis.<sup>13,14</sup> Bone tissue healing involves a complex signaling cascade, as shown in Fig. 1.<sup>15</sup> However, when the defect size is extremely large (*i.e.*,  $\geq 2.5$  cm), the natural healing mechanism is insufficient to fully repair the defect.<sup>14,16</sup> Such defects persist for the remainder of the patient's life and are known as “critical-size defects” (CSDs).<sup>17,18</sup> When the size of a bone defect exceeds the CSD threshold, it cannot repair itself, thus requiring clinical intervention.<sup>19</sup> However, despite the high incidence of bone injury, treatment selection remains controversial.<sup>20</sup> Bone defect repair involves a complex process of regeneration and

reconstruction and includes structural and functional repair.<sup>21</sup> Bone healing is a dynamic and continuous process that is accompanied by an alternating metabolic model. Currently, the most common clinical treatment for severe bone repair is bone grafting.<sup>22</sup> Bone grafts usually refer to natural or artificial materials that have positive therapeutic effects on bone regeneration.<sup>23</sup> They include autografts, allografts, xenografts, and grafts of synthetic bone materials.<sup>24</sup> In general, autografts are considered to be the gold standard for bone transplantation owing to their excellent osteointegration, biocompatibility, osteoinductivity, osteoconductivity and osteogenic properties.<sup>25,26</sup> However, autograft transplantation is an expensive



**Fig. 1** Process and mechanism of bone healing. (a) Inflammation phase; (b) bone formation phase; (c) remodeling phase. Bone healing is a dynamic and continuous process accompanied by an alternating metabolic model. In each phase, different cells and cytokines play the dominant roles.<sup>19</sup> Reproduced from ref. 19 with permission from Elsevier, copyright 2021.

<sup>a</sup>State Key Laboratory of Oral Diseases, National Clinical Research Centre for Oral Diseases, West China Hospital of Stomatology, Sichuan University, Chengdu, 610041, China. E-mail: [liaojinfeng.762@163.com](mailto:liaojinfeng.762@163.com)

<sup>b</sup>Department of Orthopedics, Orthopedic Research Institute, West China Hospital, Sichuan University, Chengdu, 610041, China. E-mail: [dove-baker@126.com](mailto:dove-baker@126.com)



procedure that requires a second operation, which can lead to new trauma as well as infection, inflammation, and chronic symptoms.<sup>27,28</sup> In addition, autografts are limited by scarcity of bone sources.<sup>21</sup> Allografts are most commonly used as an alternative to autografts, but they also have certain drawbacks, such as donor scarcity, immune rejection, poor osseointegration, and spread of infectious disease.<sup>29,30</sup> In addition, long-term use of immunosuppressant drugs is required when immune rejection occurs in allografts; this can lead to several life-threatening complications.<sup>31</sup> Therefore, the natural bone supply used for grafts cannot meet the increasing patient demand.<sup>32</sup> Numerous bone scaffolds have been developed *via* bone tissue engineering (BTE), and their application has shown great promise for bone repair.

Due to the limitations of bone grafts, BTE has been widely used for repairing bone tissue defects.<sup>33</sup> Langer and Vacanti reported that BTE technologies are the result of an interdisciplinary approach that utilizes biological and engineering principles to develop viable alternatives for repairing, maintaining, or improving bone tissue function.<sup>34</sup> BTE has the advantages of high modifiability, low risk of infection, good biocompatibility and no evident complications.<sup>1,35</sup> BTE has been used to develop biomaterials with excellent properties that promote bone regeneration and maintain and improve tissue function.<sup>36</sup> In addition, engineered bone provides a suitable environment for cell adhesion, migration, proliferation, and differentiation.<sup>37</sup> BTE represents a new strategy for inducing bone regeneration by combining biological technologies and biomaterials, which is a challenging and complicated process.<sup>38</sup> In general, BTE involves repairing bone defects by grafting scaffolds into bone defects and then gradually replacing scaffold materials with newly formed bone.<sup>39</sup> BTE approaches mainly consist of three basic elements: bone scaffolds, bone cells and growth factors, as shown in Fig. 2.<sup>40,41</sup> Scaffold materials can be divided into inorganic materials, natural or synthetic polymers, and composites.<sup>42</sup> Among all biomaterials, hydrogels are widely used as BTE scaffolds owing to their desirable properties.<sup>43–45</sup>

Hydrogels are three-dimensional (3D) networks composed of hydrophilic polymeric chains or low-molecular-mass gelators with high water content. This composition indicates that hydrogels can absorb large amounts of water, causing swelling without dissolving, and can therefore be used to fabricate extracellular matrix (ECM) simulation scaffolds.<sup>46–48</sup> High water content facilitates the diffusion of intercellular molecules and supports the growth, proliferation, and migration of bone cells.<sup>49</sup> In addition to their excellent biological properties and high water content, hydrogels exhibit good biocompatibility and facilitate inherent cell-to-cell interactions.<sup>43,50</sup> The porous structure of hydrogels allows fluid flow and the reconstruction of new blood vessels, which facilitates the diffusion of oxygen, nutrients, and metabolic waste.<sup>51–53</sup> Moreover, hydrogels can be designed in any size or shape and can be injected into bone defects to accommodate irregular shapes.<sup>54,55</sup> Owing to these excellent biological properties, hydrogels have been widely studied in tissue engineering, drug transport, and medical device research.<sup>56–62</sup> In addition, hydrogels are applied in wound dressings,<sup>63–65</sup> electronic equipment,<sup>66</sup> biosensor

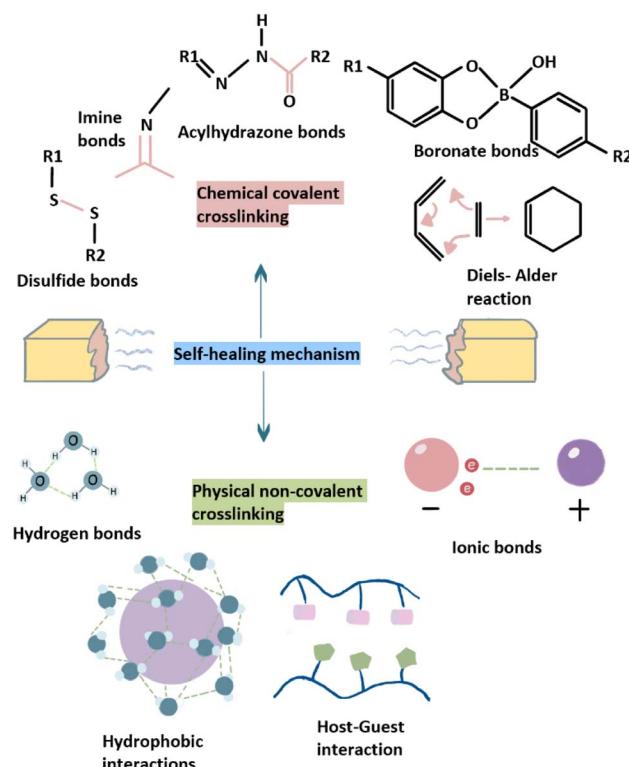


Fig. 2 The mechanisms of SHHs include chemical covalent crosslinking and physical non-covalent crosslinking.

development,<sup>67–69</sup> cell imaging,<sup>70</sup> waste disposal,<sup>71</sup> and artificial skin-like materials.<sup>72–74</sup> However, a lack of toughness, low mechanical strength and batch variation have limited the application of hydrogels in the medical field.<sup>75–77</sup> Adding inorganic and/or organic fillers to hydrogels can improve their properties and broaden their application range.<sup>78</sup> Furthermore, they cannot undergo self-healing, making it difficult to resist damage caused by fatigue or corrosion of scaffolds in human tissue.<sup>79</sup> Therefore, the integration of self-healing behavior into scaffold materials is an innovative strategy for bone defect repair.<sup>80</sup>

A self-healing hydrogel (SHH) is a special type of hydrogel that can repair itself following external damage. SHHs exhibit better fatigue resistance, reusability, hydrophilicity and responsiveness to environmental stimuli than traditional hydrogels and are therefore more suitable for regenerative medicine.<sup>81</sup> Originally, the concept of self-healing was inspired by the healing processes of natural tissues, such as the formation of bone callus, repair of broken ends after fracture, and formation of small wounds during skin surface healing.<sup>82</sup> Self-healing is a unique feature of biological systems that can restore integrity and prolong life after being damaged,<sup>83</sup> and most scaffolds used for BTE cannot self-heal, especially in wet environments. However, an ideal self-healing polymer system should repair damage autonomously in any location.<sup>83</sup> The self-healing ability of hydrogels mainly depends on the reversibility of crosslinking.<sup>84</sup> The type and number of bonds involved in crosslinking between molecules determines the properties of



SHHs (such as high mechanical strength and self-healing). Some hydrogels can add a self-healing agent to a pre-existing microporous structure. The self-healing agent is then released in response to damage to initiate self-healing at the defect site.<sup>85</sup> Moreover, SHHs can be divided into chemical covalent cross-linking and physical noncovalent crosslinking based on the specific self-healing mechanisms involved.<sup>86</sup> The dynamic covalent interactions in SHHs usually require the application of external stimuli, such as pH, alternating current, or ultraviolet light to induce hydrogel self-repair.<sup>87,88</sup> In contrast, autonomous SHHs usually use noncovalent interactions, either alone or in combination with other interactions, and do not require external stimuli, such as ionic bonding, hydrogen-bonding, supramolecular interactions, hydrophobic bonding, molecular diffusion, and chain entanglement.<sup>89</sup> Regardless of the mechanism, SHHs should be designed to match their intended application. Ideally, SHHs should respond autonomously, rapidly, and repeatedly to induce damage under mild conditions, thereby ensuring long-term use and stable function.<sup>90,91</sup> In other words, SHHs can convert irreversible damage to reversible damage.<sup>92</sup> Moreover, SHHs can maintain their original properties, such as mechanical properties, fluidity, biocompatibility, after self-healing.<sup>93</sup> Compared with hydrogels based on dynamic covalent crosslinking, SHHs based on physical noncovalent crosslinking exhibit poor mechanical properties due to weak and reversible noncovalent interactions. The self-healing ability of hydrogels is usually inversely proportional to their mechanical strength, which is generally achieved *via* enhanced crosslinking.<sup>94</sup> The mechanical properties of SHHs with multiple crosslinking, such as (nano) composites and hybrid and interpenetrating polymer network hydrogels, can be improved to some extent.<sup>65</sup> However, traditional hydrogels are generally not sensitive to external stimuli and can break without recovery. In such cases, the lifetime of hydrogels is greatly shortened, consumption of raw materials is increased, and hydrogel durability is reduced. Consequently, the use of SHHs can reduce replacement costs, and improve system safety.<sup>95</sup> Therefore, the application of traditional hydrogels in biomedicine and other fields is limited, whereas SHHs show great potential for future applications.<sup>96</sup>

At the beginning of this review, we explained how the self-healing ability of bone tissue is affected by various risk factors. Artificial surgical intervention is needed when the size of a particular bone defect exceeds the CSD threshold. Autografts are the “gold standard method;” however, owing to their limited application, BTE has attracted increasing attention. Because of their good biocompatibility and hydrophilicity, hydrogels have been widely used in studies on bone tissue repair. Although hydrogels have many advantages over other scaffold materials, they also have limitations such as insufficient toughness and low mechanical strength.<sup>97</sup> Thus, this review demonstrates that the development of SHHs is promising in the field of bone tissue repair. This review aimed to systematically discuss the wide application of SHHs in bone tissue and broaden our understanding of their prospects and challenges.

## Characteristics of SHHs

The properties of hydrogels vary greatly depending on the source of raw materials, crosslinking method, polymerization method, electric charge, environmental response, and degradability. Herein, we discussed the common properties of hydrogels and their evaluation.

### Pores and porosity

The porosity of hydrogels is beneficial for ECM secretion, cell binding, migration, and inward growth, which play an important role in directing tissue formation and function.<sup>98</sup> Currently, many techniques, including solvent casting/particle leaching, freeze-drying, gas foaming, and electrospinning technologies, can accurately control the aperture size.<sup>99</sup> Moreover, recent studies have reported optimum pore size ranges for different cells or tissue types. For example, pore sizes of  $\sim 5\text{ }\mu\text{m}$  are used for neovascularization,  $5\text{--}15\text{ }\mu\text{m}$  for fibroblast ingrowth,  $\sim 20\text{ }\mu\text{m}$  for hepatocyte ingrowth,  $20\text{--}125\text{ }\mu\text{m}$  for skin regeneration,  $70\text{--}120\text{ }\mu\text{m}$  for chondrocyte ingrowth,  $40\text{--}150\text{ }\mu\text{m}$  for fibroblast binding,  $45\text{--}150\text{ }\mu\text{m}$  for liver tissue regeneration,  $60\text{--}150\text{ }\mu\text{m}$  for vascular smooth muscle cell binding,  $100\text{--}300\text{ }\mu\text{m}$  for bladder smooth muscle cell adhesion and ingrowth,  $100\text{--}400\text{ }\mu\text{m}$  for bone regeneration, and  $200\text{--}350\text{ }\mu\text{m}$  for osteoconduction, depending on the porosity and scaffold material used.<sup>100</sup> Lan *et al.* constructed a high-porosity polyethylene glycol diacrylate (PEGDA) hydrogel foam and varied its porosity from 25% to 75% by adjusting the initial air-to-solution volume ratio.<sup>101</sup> The hydrogel foam with the highest porosity ( $\sim 75\%$ ) showed the greatest water uptake along with good water absorption, moisture retention, and exudate management.

### Swelling properties

The osmotic driving force generated by the pores in a hydrogel (through capillary forces) allows it to absorb up to thousands of times of its dry weight; thus, it can expand without dissolving until equilibrium is reached. The amount of free and bound water can be determined using various methods, such as nuclear magnetic resonance, differential scanning calorimetry, X-ray powder diffraction, dielectric relaxation spectroscopy, quasielastic neutron scattering, infrared spectroscopy, and diffusion or adsorption.<sup>102</sup> The swelling of hydrogels allows them to change their volume in response to different physical, chemical, or biological stimuli, which is critical for their self-healing behavior. Moreover, the swelling rate indicates the rate of exchange of nutrients and metabolites.

### Mechanical properties

The elastic modulus measurements of the brain, muscle, and bone tissues have been reported as 1, 10, and 100 kPa, respectively.<sup>103</sup> As a scaffold structure to simulate the ECM, the mechanical properties of hydrogels should meet the demands of practical biomedical applications and support cell activities and functions. An increase in porosity will decrease the load-bearing material per unit volume, which in turn affects the



mechanical properties of hydrogels. Gerecht *et al.* conducted a series of tensile and compression tests in poly(glycerol-*co*-sebacate)-acrylate (PGSA) and revealed that increased porosity had a substantial impact on the mechanical properties of PGSA but did not reduce ultimate strain.<sup>104</sup> In addition, the mechanical strength of hydrogels can be controlled by the nature of the network. The elastic modulus strongly depends on the type of structure (*i.e.*, crosslinking, association, entanglements) or interactions present in the network. Hydrogels with good mechanical properties can be induced by helicoidal structures and the presence of glassy nodules or crystalline domains. In contrast, hydrogels formed *via* hydrogen bonding; molecular specific binding; metal-ligand coordination; ionic, hydrophobic, host-guest, and antigen-antibody interactions;  $\pi$ - $\pi$  stacking; chain entanglements, thermal- or pH-induced gelation; and protein or polypeptide interactions have weak mechanical strength.<sup>105</sup> The mechanical properties of hydrogel biomaterials are primarily defined using theoretical frameworks of time-independent rubber elasticity and time-dependent viscoelasticity to analyze the structure of hydrogels and estimate effective crosslinking density. Currently, the most common evaluations of mechanical properties involve tensile testing for the characterization of elastic behavior and dynamic mechanical analysis for evaluating viscoelastic behavior.<sup>106</sup> Further, the less common mechanical tests include bulge testing, spherical ball inclusion, and micropipette aspiration.<sup>107</sup>

### Biological properties

The goal of evaluating the biocompatibility of any material is to determine whether it has any toxic effects on the body.<sup>108</sup> Biocompatibility is an indispensable property of hydrogels and can be defined as the ability of a material to remain in contact with organs without causing damage to surrounding tissues and without triggering any undesirable response.<sup>109</sup> A recent study evaluated the biocompatibility of a Man/BSA MeHA hydrogel using the resazurin cell viability test, alkaline phosphatase activity assay, and immunohistochemistry test; they revealed that it showed good biocompatibility with primary human alveolar bone cells.<sup>110</sup> In another study, the researchers investigated the biocompatibility of a hydrogel by evaluating the cytotoxicity of different O-HACC/polyvinyl alcohol (PVA) compound ratios and GO dosage hydrogels in normal murine fibroblasts. Their results showed that GO and chitosan (CS) quaternary ammonium salt had good biocompatibility with murine fibroblasts and were nontoxic.<sup>111</sup>

### Biodegradability

Biodegradability is another property that is essential for the design and application of hydrogels. In addition to facilitating cell proliferation and vascular infiltration, biodegradability can reduce the possibility of inflammatory reactions caused by degradation products. The degradation rate of a hydrogel should match the growth rate of the corresponding target tissue. *In vitro* cell studies have shown that if hydrogels cannot degrade over time, the morphology, proliferation, and migration of cells will be inhibited.<sup>107</sup> However, biomaterials that

biodegrade rapidly may not be able to serve as a space-filling scaffold capable of supporting new tissue formation.<sup>112</sup> Raza *et al.* discussed the influence of hydrolytic degradation in thiol-norbornene hydrogels synthesized *via* thiol-norbornene photo-click reactions on cell function by monitoring the viability of human mesenchymal stem cells (hMSCs). Their results showed that cell viability was more strongly promoted in the presence of nonhydrolytically labile hydrogels.<sup>113,114</sup> In addition, Khetan *et al.* demonstrated that the differentiation of hMSCs in covalently crosslinked hydroxyapatite (HA) hydrogels is regulated by the generation of degradation-mediated cellular traction, independent of cell morphology or matrix mechanics.<sup>115</sup>

### The self-healing ability of SHHs

In addition to the abovementioned general properties, SHHs applied under physiological conditions exhibit rapid self-healing ability.<sup>116</sup> The self-healing ability of hydrogels can be evaluated using both qualitative and quantitative methods. Qualitative methods mainly refer visual or microscopic examination to determine the healing of hydrogel wound systems. These methods involve the assessment of variables such as the degree of crack closure and the degree of broken end connection. Currently, the procedure of the commonly used method for qualitatively evaluating the self-healing ability of hydrogels is as follows. Briefly, two hydrogels each of the same size are stained with methyl orange and methyl blue. They are then sliced into two equal halves.<sup>117</sup> Next, two differently stained halves are combined in a single mold and placed at a constant temperature for a predetermined time. The self-healing process is then recorded using a digital camera.<sup>118</sup> During this process, the hydrogel is sealed in a storage bag to prevent the interference of evaporation with healing.<sup>119</sup> The self-healing ability is assessed by determining whether key properties (such as mechanical strength and elastic modulus) are restored to original levels following hydrogel repair. These measurements may include the results of compressive, tensile, or three-point bending tests. The degree of self-healing ability is represented by the healing efficiency (HE), which is calculated as follows: HE = (properties of hydrogels after healing)/(properties of the original hydrogel)  $\times$  100% (here, measurements of the properties must be performed under identical conditions). The self-healing ability of hydrogels damaged *via* shear force can be measured using cyclic rheological tests of low and high oscillatory strains. In addition, the self-healing ability may be related to the time of material in the defect. A faster healing speed is more conducive to good clinical prognoses, but different tissues possess different intracellular environments, causing variations in hydrogel performance. It is therefore necessary to select appropriate methods to quantify self-healing ability. For example, self-healing materials in dentistry should be subjected to dynamic testing conditions to simulate mastication, which would have practical clinical significance.<sup>120</sup> Moreover, SHHs should be injectable, and shear-thinning behavior is the main determinant of the injectability of hydrogels.<sup>121,122</sup> In addition to rapid healing ability, biocompatibility, and good injectability, SHHs should possess adjustable mechanical properties.<sup>123</sup> For



example, sufficient tensile and compressive strength is required to withstand stress and deformation in the environment conditions.<sup>124</sup>

### Mechanisms of self-healing hydrogels

In SHHs, self-healing can either be induced by external stimuli or interactions within the hydrogel (such as dynamic chemical bonds and noncovalent interactions).<sup>125</sup> External stimulus-induced self-healing mechanism is caused by the release of self-healing agents in microcontainers.<sup>126</sup> The main disadvantage of this mechanism is that the self-healing agent is gradually consumed during the self-healing process. Thus, it can be used for a limited number of times and cannot be repeated indefinitely.<sup>127</sup> Accordingly, autonomous self-healing has become a hot research topic in recent years owing to the diversity of potential autonomic interactions.<sup>128,129</sup> Early SHHs mainly release healing agents *via* microcapsules or microtubules, but these methods can only achieve sufficient healing once or twice.<sup>130</sup> To address these limitations, constitutionally dynamic chemistries, including noncovalent and dynamic covalent chemistries, have recently been used to construct SHHs that are capable of multiple rounds of reversible healing.<sup>131</sup> Thus, SHH design is divided into two main categories: dynamic covalent and noncovalent interactions.<sup>132</sup> The mechanism of the dynamic covalent interaction is to restore the original state by triggering a reversible process through external stimulation or a healing agent. The noncovalently bonded system is the most effective form of SHH as it can undergo self-repair without external stimulation and can fully return to the original structure and function. Noncovalent interactions are based on weak sacrificial links (such as hydrogen, ionic, or hydrophobic bonds).<sup>133</sup> These two common forms of cross-linking bonds in SHHs are shown in Fig. 2.

**Noncovalent interactions.** Noncovalent forms of cross-linking include hydrogen bonds, hydrophobic interactions, host-guest interactions and ionic bonds.<sup>134,135</sup> First, hydrogen bonds are ubiquitous and are detected in base pairs and water molecules. SHHs based on hydrogen bonds exhibit the property of rapid self-healing, which is facilitated by extremely fast breakage and connection between hydrogen bonds, *i.e.*, in the range of subpicosecond to picosecond.<sup>136</sup> Hydrogen bonds are formed when a positively charged hydrogen atom establishes an electrostatic link with a negatively charged ion.<sup>137</sup> The strength of the hydrogen bond depends on the negative charge of the ion. However, the strength of hydrogen bonds (4–120 kJ mol<sup>−1</sup>) is much lower than that of ionic and covalent bonds.<sup>138</sup> When many hydrogen bonds act together, they can still contribute significantly to the mechanical properties of hydrogels. Second, hydrophobicity occurs when water molecules form cage-like structures around hydrophobic molecules or when hydrophobic reactions occur due to host-guest interactions or formation of micelles or hydrophobic moieties. A commonly used host molecule that can be employed for this purpose is cyclodextrin (CD).<sup>139</sup> CDs are cyclic structures composed of repeating units of D-glucopyranoside linked by a 1,4-glycosidic bond. CDs act as a host with other guest molecules to form

host-guest supramolecules (HGSMS), which can then be grafted onto the polymer chain to facilitate host-guest interactions.<sup>91,140</sup> When hydrogels are damaged, HGSMS are broken down into dangling free host-guest molecules that can immediately recognize each other, recombine, and repair the damaged surface, resulting in rapid healing of the hydrogel.<sup>129</sup> Using this method, SHHs can undergo multiple damage-healing cycles and still maintain their original properties. Owing to these excellent properties, they are widely used in tissue regeneration, electronic skin and soft robotics.<sup>141,142</sup> Third, ionic bonds are formed *via* reversible electrostatic interactions between oppositely charged moieties.<sup>143</sup> Noncovalent interactions can also produce hydrogels with rapid healing time and high self-HE, but these properties are limited by their inelasticity and mechanical weakness.

**Covalent interactions.** Covalent interactions mainly involve chelation, dynamic covalent interactions, disulfide bonds, boronate ester bonds and Diels–Alder (DA) reactions.<sup>144,145</sup> In chelation, a covalent bond is formed by a coordination bond between a ligand and positively charged transition metal ion. Compared with covalent bonds, chelating bonds have high adhesion, elasticity, and reversibility.<sup>146</sup> By designing dynamic covalent bonds, polymer hydrogels can be intrinsically undergo self-healing in response to external stimuli, such as pH, light, or temperature.<sup>147,148</sup> However, most polymer hydrogels do not undergo self-healing or -repair after damage because they contain nondynamic covalent bonds.<sup>149</sup> The difference between dynamic and standard covalent bonds is that dynamic covalent bonds have both the stability of covalent bonds and reversibility of noncovalent bonds. Imine bonds were first discovered by the German chemist Hugo Schiff; therefore compounds based on the imine bond are termed as “Schiff bases.” An imine bond is a strong covalent bond formed by the reaction of an aldehyde with a primary amine.<sup>150</sup> Moreover, imine bond formation is a dehydration process that can be performed under mild conditions.<sup>151</sup> Schiff base reactions usually occur between electrophilic carbon (such as ketones or aldehydes) and nucleophilic substances (such as amine or hydrazine) to form a reversible covalent imine or hydrazone bond.<sup>82</sup> For example, hydrogels *via* dynamic Schiff base reaction between amino groups from quaternized CS and aldehyde groups from benzaldehyde-terminated Pluronic®F127 (PF127-CHO) polymer have been shown to promote functional healing of the tendon–bone interface.<sup>58,152</sup> Disulfide bonds are covalent bonds that are indispensable for the 3D structure of many proteins. The formation of disulfide bonds is related to the amino acid cysteine (Cys). The Cys side chain has a reactive sulfhydryl group that can be easily oxidized to form disulfide bonds. This can be used in hydrogel preparation; for example, Chen *et al.* reported a mercapto-disulfide bond exchange strategy for constructing dynamic covalent crosslinking keratin hydrogels.<sup>145</sup> Boronate ester bonds formed by combining diols and boronic acid can form boronate bonds.<sup>177</sup> Hydrogels with such bonds exhibit efficient self-healing and mechanical strength but are sensitive to pH changes. Bond formation occurs only when the pH is greater than or equal to the pK<sub>a</sub> value of boric acid (*i.e.*, usually >8 pK<sub>a</sub>).<sup>178</sup> However, most tissues in the body operate at



a neutral pH, and many cells die when the pH exceeds 8. In other words, boronate ester-based hydrogels are rarely compatible with physiological conditions.<sup>179</sup> Recent studies have shown that boronic acid-based hydrogels undergo self-healing at neutral and acidic pH. For example, hydrogels crosslinked *via* diol complexation with 2APBA units exhibit self-healing behavior at lower pH levels.<sup>180</sup>

The “click chemistry,” also referred to as “link chemistry,” “dynamic group chemistry” or “rapid combinatorial chemistry,” has many salient advantages, including simple reaction conditions, fast reaction speed, nontoxic and harmless secondary products, low generation of impurities, and high yield.<sup>151</sup> The DA reaction, which belongs to a class of biocompatible reactions designed to bind substrates to specific biomolecules, is also important for click chemistry.<sup>181</sup> DA reactions have also been used to construct SHHs that react with diene and dienophile reagents.<sup>182</sup> However, DA reactions require high temperatures (>100 °C) to induce reversibility. This high temperature requirement limits the biomedical application of DA reactions.

Based on the above discussion, we can clearly understand the mechanism and importance of the self-healing effect of hydrogels. According to different intramolecular interactions involved in SHHs, their mechanisms of action can be classified as noncovalent or covalent interactions. As different SHHs act in different ways and have varying self-healing mechanisms, all of them are not perfect (Table 1).

## Application of self-healing hydrogels for bone defect repair

Many recent studies have reported the preparation of high-performance, SHHs that possess good histocompatibility, adjustability, and nontoxicity. The use of SHHs in bone defect repair is therefore an area of active research. For example, Wu *et al.* performed thiol modification by introducing disulfide crosslinking to NIPAAm-g-CS (NC) hydrogels and assessed their toxicity and bioactive effect using *in vitro* cell experiments.<sup>156</sup> They revealed that the hydrogel had no cytotoxicity. In addition, they solved the problem of poor biocompatibility and biodegradability by copolymerizing NC with NIPAAm during hydrogel formation. The original NC hydrogel has been shown to be a suitable cell carrier scaffold owing to its good biocompatibility and biodegradability. However, its use for biomedical applications was limited by its weak mechanical properties. Wu *et al.* artificially enhanced the mechanical properties of NC hydrogel, incorporated thiol side chains into CS, and formed disulfide bonds through thioloxidation, thus completing the modification of the NC hydrogel. This hydrogel is therefore expected to be a cell-bearing tissue regeneration biomaterial. In another study, Lu *et al.* used acylhydrazone-based crosslinking with or without *in vivo* DA cross-linking to produce dynamic SHHs. These hydrogels were loaded with bone morphogenetic protein-4 and injected into bone defects to promote bone regeneration. Ma *et al.* prepared an injectable hydrogel consisting of alginate oxide hybrid HA nanoparticles (NPs) and carboxymethyl CS *via*

Schiff base reaction. The self-healing properties of these hydrogels were verified *via* splicing and rheological experiments (Fig. 3). These hydrogels have broad application prospects for BTE.<sup>183</sup> Pan *et al.* prepared a novel biocompatible injectable and self-healing nanohybrid hydrogel *via* reversible Schiff base reaction between  $-\text{HC=O}$  of oxidized sodium alginate (OSA) and  $-\text{NH}_2$  of glycol CS mixed with calcium phosphate (CaP) NPs. The results of their experiments showed that this novel hydrogel is an ideal candidate for BTE applications and drug delivery.<sup>184</sup> Bai *et al.* prepared a self-healing dual crosslinked injectable hydrogel using DA click chemistry for repairing skull defects in rats.<sup>185</sup> In their experiment, they first used maleimido terminated F127 (F127-AMI) and furfurylamine-grafted chondroitin sulfate (ChS-furan) to synthesize F127-crosslinked ChS (F127@ChS) *via* DA click chemistry. Next, the dual crosslinked hydrogels were prepared based on F127@ChS and PEG-AMI. Shi *et al.* proposed a strategy for the assembly of silk fibroin (SF)-based hydrogels under physiological conditions based on dynamic metal-bisphosphonate (BP) coordination bonds between SF microfibers (mSF) and a polysaccharide binder.<sup>125</sup> They used biominerization to generate mSF coated with CaP and chelated by a bisphosphonate ligand of the binder to form a reversible crosslink (Fig. 4). Based on the reversibility of the coordination bond between CaP and BP ligands, the SF-based hydrogels exhibited self-healing properties and did not require external stimulation during healing. In addition, these hydrogels had shear thinning properties that allow them to fill irregularly shaped tissue defects without breaking. However, these hydrogels are characterized by poor mechanical properties and insufficient stability under physiological conditions. To overcome this limitation, photosensitive polyacrylate groups were introduced into the adhesive to improve the mechanical properties of self-healing SF-based hydrogels when exposed to UV light. To demonstrate this phenomenon, the authors implanted a composite scaffold prepared using SF into rat skulls with a severe defect (diameter = 8 mm) and examined the animals 4 and 8 weeks after implantation. They concluded that the hydrogel stimulated the formation of new bone in the CSD rat skull model. In addition to single covalent crosslinking, two different covalent crosslinking methods have been used to form high-performance SHHs. In a previous study, Lu *et al.* attempted to overcome the limitations of hydrogels based on ChS (which show insufficient strength and inaccurate mechanical tunability and are non-self-healing and noninjectable) using DA click chemistry and dynamic hydrazone bond crosslinking to form hydrogels with excellent performance. Compared with hydrogels formed *via* single crosslinking, the abovementioned hydrogels showed increased viability and reduced apoptosis in rat MSCs as well as excellent tissue adhesive ability *in vivo*.<sup>186</sup> These experimental results demonstrated that the hydrogels were suitable for use as scaffolds in rat skull tissue engineering. In this experiment, new bone tissue formation was detected in the skull defect area.

In recent years, hydrogels based on noncovalent interaction have been widely investigated. Some experimental methods for preparing high-performance hydrogels *via* noncovalent cross-linking are listed below. Bai *et al.* fabricated a novel inorganic-



Table 1 Advantages and disadvantages of SHHs and their applications

Classification	Self-healing mechanisms	Advantages	Disadvantages	Properties	Examples	Applications	References
Chemical covalent cross-linking	Imine bonds (Schiff base)	Excellent mechanical strength	Amadori rearrangement	Injectability, fast gelation process, super hydrophilicity, excellent biocompatibility, excellent neutral stability and ultrasensitive pH-responsive behavior	AHA/cystamine dihydrochloride (AHA/Cys) hydrogels, the double-network GMO hydrogels (DN-GMO), QCS/OHA-PEDOT-BBH-EGF hydrogels	Cancer treatment, controlled drug release, biosensors, wound dressings	<sup>128</sup> and <sup>153-155</sup>
	Disulfide bonds	Room temperature interaction	Produce reaction by-products (mercaptan, toxic)	Enhanced mechanical properties, good biocompatibility, exceptional stretchability, complete and rapid degradation	Dynamically crosslinked gold nanoparticle-hyaluronan hydrogels, OSA-HPCS hydrogel, PVA/ce-MoS <sub>2</sub> hydrogel	Controlled drug and gene delivery	<sup>156-159</sup>
	Boronate ester bonds	Nonspecific chemical bonding <i>in situ</i> formation	Bond cannot be formed under extreme pH conditions	Good mechanical properties, excellent biocompatibility and conductivity	Cellulose based hydrogels from CMC-B(OH) <sub>2</sub> and PVA, CNC- <i>g</i> -PGMA/ PDMA- <i>stat</i> -PAPBA NC hydrogels, glucose-responsive hydrogel electrode, poly(NIPAAm- <i>co</i> -APBA- <i>co</i> -AAm) and PVAd	Tissue engineering, wound repairing, controlled drug release, detection of glucose	<sup>160-162</sup>
	DA “click chemistry”	One-step reaction, thermally reversible reaction	Requires high temperature and a long time	Good biocompatibility, biodegradability, excellent gel-forming properties	F127@ChS-PEG-OChS hydrogel, self-assembly and core crosslinking of PFMA- <i>b</i> -PSS and PFMA- <i>b</i> -PMTAC and formation of a hydrogel	Drug delivery, cartilage tissue engineering	<sup>163-165</sup>
Physical noncovalent cross-linking	Hydrogen bond	Nontoxic chemical bonding	Poor mechanical strength	High stability, biocompatibility and poor mechanical strength	Polydopamine-polyacrylamide (PDA-PAM) single network hydrogel, TTA-UC hydrogel, the urea-PA hydrogel	Detection of drug release, antibacterial activity	<sup>135</sup> and <sup>166</sup>
	Hydrophobic interaction	Widely formed in every water-containing system	Poor bonding strength	Conductive, enhanced mechanical properties tunable mechanical properties and thermal response behavior	Stimuli-responsive graphene-based hydrogel, the MA-UPy-PEO-UPy-MA hydrogel, SHH based on the hydrophobic interaction of a biocompatible four-arms star polymer, poly(ethylene glycol)- <i>b</i> -poly( $\gamma$ - <i>o</i> -nitrobenzyl-L-glutamate)	Drug release, tissue engineering stents, wound dressings, and biosensors	<sup>167-169</sup>



Table 1 (Contd.)

Classification	Self-healing mechanisms	Advantages	Disadvantages	Properties	Examples	Applications	References
	Host-guest interaction	Convenient to adjust the molecular properties of the guest species through a wide range of stimuli	Specific in some cases, binding to each other temporarily	Superior self-healing, injectability, flexibility, stimuli-responsiveness and biocompatibility	Injectable poly(pseudorotaxane hydrogels, the cellulose-derived hydrogels (CAAs), the injectable Tet-Ada/poly (β-CD) hydrogels	Injectable drug depots, wound dressings, advanced inks for 3D printing of cell scaffolds	<sup>170</sup> and <sup>171</sup>
	Ionic bonds	High toughness, anti-fatigue abilities, antifreezing capability, and stretchability	Slow healing process, good thixotropic property	Biodegradability, non-toxicity, biocompatibility, renewable and easy accessibility	Ionic alginate hydrogels, Phos-cycC6/BPMoc-F3/ $\text{SO}_x/\text{Ca}^{2+}$ hydrogel, mesoporous silica/polyacrylate hybrid hydrogel, chitosan/polyzwitterion-based double network hydrogel	Drug delivery capacitive, resistive bimodal sensors	<sup>172–176</sup>

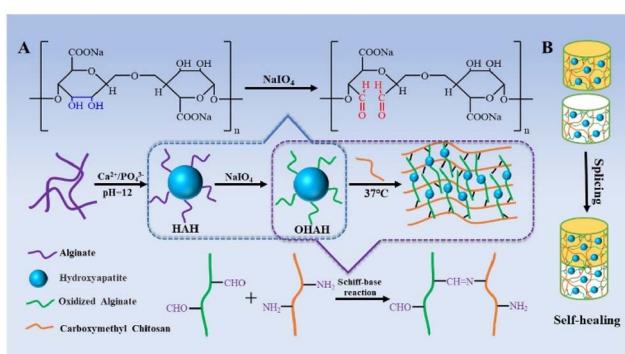


Fig. 3 Schematic illustration (A) of the preparation of injectable hydrogels via Schiff base reaction. (B) The self-healing property of the hydrogels.<sup>183</sup> Reproduced from ref. 183 with permission from Elsevier, copyright 2020.

organic hybrid hydrogel with self-healing ability based on SF using dynamic host–guest interactions.<sup>91</sup> Herein, SF was combined with β-CD and cholesterol molecules *via* standard amidation and esterification reactions to assemble supramolecular hydrogels with self-healing behavior. Next, HA NPs were added to the hydrogel to construct inorganic–organic hybrid composites (known as SF@HG@HA). Finally, a CSD model was used to evaluate the ability of this hydrogel to promote the formation of new bone. It was concluded that the hydrogel showed good biocompatibility and biodegradation and could promote the formation of new bone in an *in vitro* cell experiment. To enhance the mechanical properties of hydrogels, many studies have focused on the preparation of multinetwork SHHs. For example, Bi *et al.* prepared a CS-PVA double-network (DN) hydrogel that exhibited both high strength and toughness. This hydrogel was based on multihydrogen bond interactions using a freezing–heating alternating treatment, which was

applied to an alkaline solution of CS-PVA.<sup>187</sup> Their experiments showed that the hydrogel preparation process was simple, nontoxic, and harmless and may be suitable for use in tissue engineering repair. In addition, the adhesion of macrophages to the material matrix plays a crucial role in the implantation of

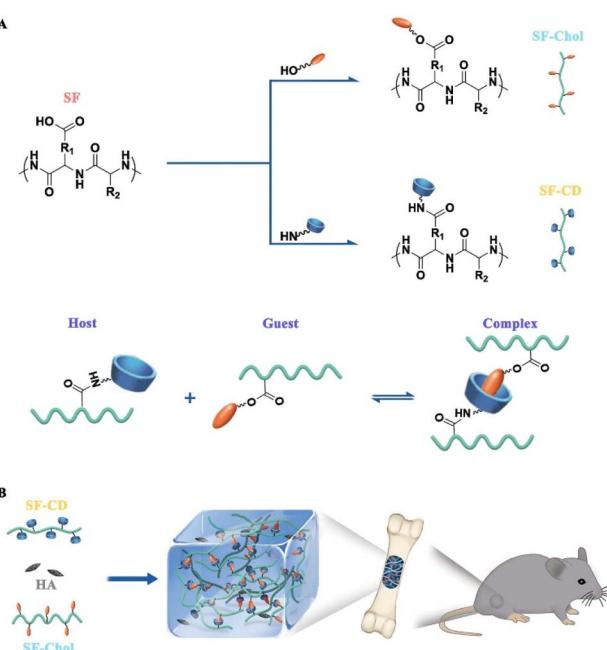


Fig. 4 Construction of the novel bone grafts (SF@HG@HA) with self-healing capability. (A) Synthesis of specific host (SF-CD) and guest (SF-Chol) macromers. Interaction of β-cyclodextrin (CD, host) and cholesterol (Chol, guest) in formation of a reversible host–guest (HG) complex. (B) Schematic of supramolecular hydrogel formation through host–guest complexation and its application as bone graft for promoting bone regeneration.<sup>91</sup> Reproduced from ref. 91 with permission from Elsevier, copyright 2021.



biomaterials into the human body and the application of biomaterials in specific biomedical processes. Xu *et al.* enhanced the adhesion of macrophages by improving the hydrophobicity of the surface of a methyl-gellan gum hydrogel.<sup>188</sup> It has been demonstrated that the hydrophobicity of a substrate can be used to regulate the macrophage response; this property can be beneficial for wound healing or repair. Feng *et al.* prepared gelatin macromolecules *via* host-guest interactions between aromatic residues of gelatin and free-diffused photocrosslinked acrylic  $\beta$ -CD monomers.<sup>189</sup> Subsequent macromolecules were crosslinked with each other to produce highly elastic supramolecular gelatin hydrogels that were only crosslinked *via* weak host-guest interactions (Fig. 4). The hydrogel thus obtained showed the following advantages. First, it maintained excessive compression and tensile strength. Second, it showed fast self-healing after mechanical damage. Third, it could be injected in a gel state and remolded to the target geometry. Fourth, it could promote cell infiltration and migration into the hydrogel. Fifth, excessive  $\beta$ -CD makes the hydrogel adhere tissue and enhances the loading capacity and delivery of hydrophobic drugs. The researchers implanted this hydrogel in rats with skull defects and conducted cell and animal studies 10 weeks later. Finally, they concluded that the hydrogel supported cell recruitment, differentiation, and bone regeneration, making it a useful biomaterial carrier for therapeutic cells and drugs *via* a minimally invasive procedure.

### SHHs for cell loading

Early cell cultures were performed using stiff materials such as polystyrene and glass (conventional 2D culture). The 2D cultures are simple to perform but cannot mimic the *in vivo* structure, and cells cultured in this environment tend to behave abnormally.<sup>190</sup> This led to the development of sophisticated 3D systems in which cultured cells are embedded in hydrogels rather than on top of substrates. Hydrogels have biological and mechanical properties similar to those of biological tissue and can mimic the native ECM.<sup>191</sup> Moreover, the degradation of hydrogels is inhibited during the growth, migration, and proliferation of cells. Yang *et al.* prepared cellulose-based SHHs *via* dynamic covalent hydrazone bonding and found that L929 cells could readily proliferate in 3D hydrogel environments.<sup>192</sup>

To balance the advantages and disadvantages of covalent and noncovalent crosslinking, many studies have focused on the preparation of DN SHHs based on two forms of crosslinking. Hydrogels are emerging as carriers to encapsulate cells and drugs.<sup>193</sup> Initially, conventional hydrogels commonly used for cell encapsulation were based on static chemical covalent crosslinking, but the resulting hydrogels lacked dynamic properties (such as injectability or self-healing).<sup>194</sup> Subsequent studies have shown that dynamic hydrogels based on physical noncovalent crosslinking can heal themselves and are suitable for injection, which compensates for the limitations of chemically crosslinked hydrogels. However, these hydrogels are less stable and biocompatible. Feng *et al.* developed unique cell-infiltrable and injectable (Ci-I) gelatin hydrogels, which are largely stabilized by physical crosslinking caused by host-guest

interactions and are further reinforced by limited chemical crosslinking.<sup>195</sup> They further evaluated the efficacy of Ci-I gelation hydrogels as cellular carriers for treating enclosed bone abnormality in steroid-associated osteonecrosis (SAON) of the femoral head. *In vivo* animal studies have shown that these hydrogels can retain their original mechanical properties after injection, promote the regeneration of bone *in situ*, facilitate cell chemotaxis and aggregation, and accelerate the healing of SAON. In addition, these hydrogels are easy to prepare and can continuously deliver hydrophobic drugs and cells. This is the first study demonstrating the feasibility of using injectable hydrogels to encapsulate stem cells and small molecules to treat bone disorders in deep and enclosed anatomical locations (*e.g.*, SAON in the hip).<sup>195</sup> Another combined physical-chemical crosslinking method has been used to prepare hydrogels with desirable properties. Demineralized bone matrix (DBM) powder is known to be a potential alternative bone graft material because of its similar composition and structure to autologous bone and its ability to promote bone regeneration.<sup>196</sup> However, the limitations of DBM, such as easy inactivation of growth factors during the preparation process, and lack of bone regeneration cells, hinder its wide application in bone transplantation. Li *et al.* introduced hypoxic-pretreated bone marrow stromal cells (BMSCs) to provide growth factors and bone regeneration cells for bone formation. Furthermore, they prepared an injectable SHH based on a double crosslinking structure, in which a dynamically crosslinked Schiff base network acted as a self-healing component that allows injection of payloads into the defect site. Moreover, a borax ion cross-linked physical network strengthened its mechanical properties.<sup>13</sup> This was used to transport DBM powder and hypoxia-pretreated BMSCs to the bone defect site. Finally, the experimental results showed that the bone defects of rabbits were almost completely healed after 12 weeks of applying hydrogel/DBM/BMSC. In addition, Deng *et al.* constructed a novel DN biocompatible hydrogel using PEGDA and short-chain CS *via* ionic-covalent crosslinking.<sup>197</sup> The CS-based ionic network and PEGDA-based covalent network as well as the hydrogen bonds between them together provide excellent mechanical properties. Good mechanical properties are one of the advantages of hydrogels in 3D printing technology. Moreover, this modified hydrogel has potential for tissue engineering because of its stronger, printable properties.

With technological development and improvements in tissue regeneration, increasing number of studies have focused on replicating or imitating the complex structure and function of tissues and organs. The application of 3D printing technology has become a research hotspot due to the high degree of formability and controllability of the printed hydrogels. The 3D printing technology is currently a promising biotechnology tool for tissue engineering.<sup>198</sup> Recently, hydrogels based on reversible noncovalent interactions have been gradually developed using 3D bioprinting. However, most noncovalent crosslinking hydrogels are fragile and can be easily damaged. Although they have self-healing ability, this process takes a long time; therefore, none of these hydrogels are suitable for 3D bioprinting. Hydrogels that can be used in 3D printing technology should



show shear thinning properties, good mechanical properties, appropriate yield strength, and fidelity. Therefore, to overcome these limitations, Zhang *et al.* constructed biomimetic scaffolds using hydrogel microparticulates as 3D bioprinting ink, which showed excellent mechanical properties and rapid self-healing under ambient conditions.<sup>199</sup> CS methacrylate (CHMA) and PVA, both known as biocompatible polymers, are hybridized to prepare hydrogels with mechanical properties that are tunable *via* the chemical crosslinking of CHMA and physical crosslinking of PVA through freeze–thawing (Fig. 5). However, the obtained CHMA/PVA hydrogels were rigid and not suitable for 3D printing. Therefore, the authors pressed the hydrogel through a nozzle of a specific diameter, and the gels were transformed into a slurry of microparticles, which became thixotropic. After centrifugal degassing, hydrogels were formed *via* hydrogen bonds between the particles. Therefore, they exhibited a typical shear-induced reversible gel–fluid transition. Such printed hydrogel scaffolds are conducive to cell adhesion and growth, and can thereby induce spheroid BMSC formation. This has broad future prospects for bone regeneration in tissue engineering in the future. Furthermore, Zhao *et al.* prepared a biomaterial of 3D printed porous metal scaffolds and infliximab-based hydrogels. This promising biomaterial had self-healing, histocompatible, and anti-inflammatory properties. Experimental results showed that the composite scaffold can repair bone defects in a rabbit model of severe rheumatoid arthritis.<sup>200</sup> It is easy to speculate that SHHs suitable for 3D printing have higher performance requirements, and when these difficulties are overcome, they will have great potential for bone repair.

Recently, the application of photothermal therapy (PTT) in bone regeneration has attracted considerable attention. PTT is a new hyperthermia method that uses the photothermal effect of different types of photothermal agents to convert absorbed light energy into heat energy, which helps facilitate efficient and

noninvasive treatment of various diseases.<sup>201</sup> PTT has great potential for both wound healing and bone regeneration by promoting MSC differentiation and osteoblast maturation.<sup>202</sup> For example, gold nanorods/nano-HA (nHA) were shown to exert photothermal therapeutic effects on postoperative tumor and bone defect repair in a mouse model of tibial osteosarcoma.<sup>203</sup> In another study using a rat skull defect model, a GelMA/PMMA/PDA hydrogel with mild PTT showed a stronger bone repair effect than pure hydrogel and control.<sup>204</sup> Luo *et al.* prepared the OSA–CS–polydopamine-decorated nHA (PHA)–DDP (cisplatin) bifunctional hydrogel with photothermal effect, which promoted the adhesion and proliferation of bone MSCs *in vitro* and further induced bone regeneration *in vivo*.<sup>205</sup> Thus, the combination of PTT with chemotherapy, photodynamic therapy, or immunotherapy may improve efficacy and reduce side effects.<sup>206</sup> Matheny performed a prospective, randomized, controlled, blinded clinical trial to evaluate the safety and efficacy of a novel self-crosslinked HA hydrogel. This hydrogel was compared with carboxymethylcellulose viscous foam in terms of promoting healing after ethmoidectomy. This study concluded that the hydrogel provided superior wound healing in this experiment.<sup>207</sup> Twelve patients requiring extraction of premolars and implants were selected for randomized controlled trials. Patients in the control group received a glass-reinforced HA synthetic bone substitute, Bonelike by Bioskin® (BL®), and those in the experimental group received DEXGEL bone. The stability of primary implants was analyzed using the implant stability coefficient method; the study finally concluded that the hydrogel reinforcement material was easy to handle and showed good defect healing effect.<sup>208</sup> Currently, only a few clinical trials have examined the application of hydrogels in bone repair, which poses a challenge for future clinical transformation.

### SHHs for drug delivery

In addition to the abovementioned applications, SHHs can be used for drug delivery, wound dressings, cell culture, and diagnostic applications (such as bioassays and bioimaging). In terms of drug delivery, hydrogel is a 3D network structure that can carry and deliver small drug molecules and absorb exudates to promote wound healing. For example, injectable hydrogels usually have a pore size of 50–300  $\mu\text{m}$ .<sup>209</sup> The speed of drug absorption is different from that of drug delivery, and using hydrogels can help overcome the common limitations of slow absorption and low efficiency of the traditional drug delivery method.<sup>210</sup> However, when the hydrogel is damaged, the concentration of drug rapidly increases to the peak concentration, which can induce toxicity in local tissues. SHHs may be a better alternative to avoid this risk because they have a rapid self-healing ability and can deliver therapeutic drugs and cells in a controllable manner.<sup>211</sup> Therefore, the effectiveness of SHHs depends on the specific drugs and cells delivered.

### SHHs for DNA delivery

Currently, the application of gene therapy in bone regeneration is an important research topic. One area of research focuses on

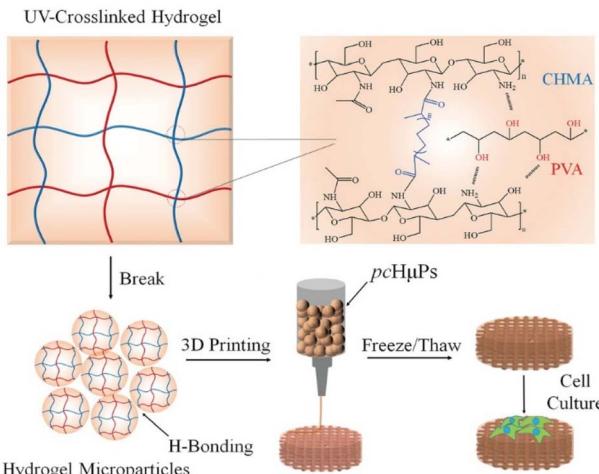


Fig. 5 Schematic illustration to the preparation of the self-healing pre-cross-linked hydrogel microparticles (pcHμPs) by 3D printing for cell spheroid growth.<sup>199</sup> Reproduced from ref. 199 with permission from Wiley, copyright 2020.



the development of polymer substrates that can deliver DNA. Among them, the most common polymer is poly(lactate-glycolic acid) (PLGA), but its degradation products are known to damage DNA.<sup>212</sup> Hydrophilic porous hydrogels that carry drugs or cells can compensate for the lack of PLGA with their high delivery efficiency. Adding DNA to hydrogels that are already widely used in bone and cartilage tissue engineering to deliver drugs or cells can help alter cell behavior and thereby enhance tissue formation.<sup>213</sup> The encapsulation of DNA *via* polymerization can control the release of DNA both spatially and temporally, thereby sustaining the local delivery of therapeutic factors for tissue regeneration.<sup>214</sup> Dadsetan *et al.* demonstrated the potential of an oligo (PEG) fumarate (OPF) hydrogel for sustained delivery of DNA complexes by using OPF hydrogels to transport DNA and bone cells.<sup>215</sup> Komatsu *et al.* found that gelatin hydrogel as a substrate for local gene delivery was more capable of inducing bone regeneration than atelocollagen.<sup>216</sup> Localized gene delivery is a promising alternative therapy as it may allow the sustained expression of specific osteoinductive growth factors in cells near the damaged site.<sup>217</sup>

## Prospective and challenges

Bone is the second most transplanted tissue in the world after blood, and traditional bone transplantation surgery has significant disadvantages, such as pain, high cost, and susceptibility to infection.<sup>218</sup> As one of the key materials in BTE, scaffolds should have many specific properties. However, the osteoinductivity of synthetic materials is lower than that of allograft materials and has therefore not been widely adopted.<sup>219</sup> However, hydrogels have been widely used as scaffolds in BTE due to their good biocompatibility, nontoxicity, and injectability. Because of the great prospects hydrogels for bone repair, many researchers have prepared synthetic hydrogels with better properties than natural hydrogels using interdisciplinary methods.

In terms of clinical translation, many facial corrections and esthetic hydrogel-based products have been approved by the US Food and Drug Administration.<sup>193</sup> Based on the clinical-trials.gov database (<https://clinicaltrials.gov/>), 514 completed and recruiting clinical trials have been performed in various application areas, including cancer treatment, esthetic correction, spinal fusion, tissue regeneration, and incontinence.<sup>220</sup> Despite the fact that SHHs are a hot topic in the field of bone repair, market penetration is low due to the lack of clinical studies demonstrating their effectiveness. This review discusses the unsolved challenges and clinical translational potential of hydrogels to facilitate the adoption of self-healing hydrogels in clinical settings. We focused on the following aspects. First, the potential side effects and long-term efficacy of hydrogel injection/implantation are uncertain. The immune response to foreign implanted biomaterials generally consists of inflammatory events and wound healing processes that lead to fibrosis. Vegas *et al.* found that the distribution of triazole modification creates a unique hydrogel surface that inhibits recognition by macrophages and fibrous deposition.<sup>221</sup> Second, hydrogels for specific diseases require

considerable care regarding the appropriate route of implantation and lowest effective dose. For instance, an inappropriate implantation route may lead to postoperative complications such as swelling, nodules, and pain. Moreover, most drugs are dose-dependent, and drug delivery-related hydrogels must balance benefits and side effects when releasing the drug. Third, self-healing hydrogels are difficult to prepare and are not yet ready for mass production. For hydrogels embedded with active substances or living cells, preservation is another challenge for large-scale clinical applications. Fourth, studies have shown that the intracellular stromal microenvironment is dynamic, and cellular behavior and the associated signaling cascades remain unexplored. When cell-loaded hydrogels are implanted into the complex microenvironment, various cells in the body as well as those in the hydrogel will interact with the surrounding cells *via* paracrine crosstalk, resulting in negative effects.<sup>103</sup> Finally, the long-standing incompatibility of SHH toughness and rapid self-repair has not yet been fully addressed.<sup>222</sup> Although some strategies have been formulated to deal with these problems, it is necessary to continue to innovate and optimize the performance of hydrogels.

The prospect of combining hydrogels with advanced biotechnology is also notable. For example, when SHHs with good tensile properties are combined with 3D printing technology, the hydrogels obtained can be arbitrarily printed into target shapes to adapt to different sizes and shapes of bone defects. Furthermore, mechanical self-healing properties can be improved by adding reinforcement materials (such as inorganic or organic fillers). Moreover, printed hydrogels can be dynamically adjusted. Dynamic adjustment involves the introduction of the fourth dimension into the 3D structure therefore, it is known as “4D printing” technology. The combination of SHHs and 4D printing technology enables hydrogels to be controlled in both time and space. Introducing the dimension of time also means that the morphology of hydrogels can change over time. This morphological adjustment property enables hydrogels to morphologically adjust according to various stimuli in the bone tissue healing process, which involves different stages of bone repair. However, even if 4D printing has superior features, it is not yet perfect. As the mechanisms involved in 4D printing technology are extremely complex, simulation of the complex dynamic deformation of the original tissue is a difficult problem, which needs further investigation on 4D printing technology.

The first market appearance of a hydrogel was reported in 1949, wherein PVA was crosslinked with formaldehyde. This hydrogel was marketed under the trade name Ivalon and was used in biomedical implants.<sup>223</sup> However, the real turning point in the production and use of hydrogels was the synthesis of poly(2-hydroxyethyl methacrylate) gels, which were invented and studied by Otto Wichterle and Drahoslav Lim during the development of modern soft hydrogel contact lenses in 1960.<sup>224</sup> This represented the starting point for the spread of a flourishing hydrogel market. Until 2016, the hydrogel market was valued at USD 10.87 billion and was projected to reach USD 15.33 billion by 2022. This shows a compound annual growth rate of 6.04% from 2017 to 2022.<sup>225</sup> Finally, biomedical research



is advancing rapidly. With the continuous creation of new advanced technologies, research on SHHs can be further refined, and the performance of SHHs will be more aligned with human needs. Over time, the unique advantages of SHHs can compensate for the limitations of autografts, thereby becoming a widely used bone repair material showing consistently good performance. However, considerable effort is required for improving SHH performance.

## Conclusions

Herein, we conducted a systematic review of available research related to the self-healing behavior, self-healing mechanism, main performance requirements and application of hydrogels in bone repair. Depending on the biomaterial from which hydrogels are prepared, their self-healing mechanisms also differ, and the properties of hydrogels vary accordingly. Moreover, there are many factors that cause bone defects, and bone repair remains a major challenge. Currently, SHHs are considered a promising scaffold material in tissue engineering, only SHHs with adequate performance requirements have been designed, owing to the differences in bone defects among clinical cases. Thus, the design and preparation of SHHs are promising but challenging, and additional research is warranted to further improve them. However, we speculate that SHHs should be used in clinical trials as soon as possible to repair bone defects in patients.

## Conflicts of interest

There is no conflict of interest for all the authors.

## Acknowledgements

This work was financially supported by the National Natural Science Foundation (32171354, 31972925, 31700839), the Fundamental Research Funds for Central Universities.

## References

- 1 E. J. Sheehy, D. J. Kelly and F. J. O'Brien, *Mater. Today Bio*, 2019, **3**, 100009.
- 2 F. M. Klenke and K. A. Siebenrock, in *Reference Module in Biomedical Sciences*, Elsevier, 2016, DOI: [10.1016/B978-0-12-801238-3.99488-1](https://doi.org/10.1016/B978-0-12-801238-3.99488-1).
- 3 A. A. El-Rashidy, J. A. Roether, L. Harhaus, U. Kneser and A. R. Boccaccini, *Acta Biomater.*, 2017, **62**, 1–28.
- 4 C. Li, J. Sun, K. Shi, J. Long, L. Li, Y. Lai and L. Qin, *J. Mater. Chem. B*, 2020, **8**, 4575–4586.
- 5 E. Mancuso, L. Shah, S. Jindal, C. Serenelli, Z. M. Tsikrteas, H. Khanbareh and A. Tirella, *Mater. Sci. Eng., C*, 2021, **126**, 112192.
- 6 J. Raphel, M. Holodniy, S. B. Goodman and S. C. Heilshorn, *Biomaterials*, 2016, **84**, 301–314.
- 7 Z. Yang, H. Tao, Z. Ye, L. Jin, N. Lin and D. Yang, *J. Int. Med. Res.*, 2018, **46**, 3219–3225.
- 8 J. F. Liao, K. Shi, Y. P. Jia, Y. T. Wu and Z. Y. Qian, *Bioact. Mater.*, 2021, **6**, 2221–2230.
- 9 P. N. Soucacos, Z. T. Kokkalis, M. Piagkou and E. O. Johnson, *Injury*, 2013, **44**, S70–S75.
- 10 V. Chadayammuri, M. Hake and C. Mauffrey, *Patient Safety in Surgery*, 2015, **9**, 32.
- 11 N. V. Shrivastava, A. K. Tiwari, R. Kumar, S. Patil, D. Tripathi and S. Badhyal, *J. Biomech. Eng.*, 2021, **143**, 081011.
- 12 R. Fleischmann, R. Landewe and J. S. Smolen, *Semin. Arthritis Rheum.*, 2016, **46**, 279–285.
- 13 D. Li, Z. Yang, X. Zhao, Y. Luo, Y. Ou, P. Kang and M. Tian, *J. Mater. Chem. B*, 2021, **9**, 479–493.
- 14 E. H. Schemitsch, *J. Orthop. Trauma*, 2017, **31**, S20–S22.
- 15 M. B. Sordi, A. Cruz, M. C. Fredel, R. Magini and P. T. Sharpe, *Mater. Sci. Eng., C*, 2021, **124**, 112055.
- 16 D. Li, Q. Hu, P. Kang, J. Yang, Z. Zhou, B. Shen and F. Pei, *International Orthopaedics*, 2018, **42**, 2787–2795.
- 17 D. Li, X. Xie, P. Kang, B. Shen, F. Pei and C. Wang, *J. Orthop. Sci.*, 2017, **22**, 1060–1065.
- 18 L. Yang, I. Ullah, K. Yu, W. Zhang, J. Zhou, T. Sun, L. Shi, S. Yao, K. Chen, X. Zhang and X. Guo, *Biofabrication*, 2021, **13**(3), 035007.
- 19 G. Zhu, T. Zhang, M. Chen, K. Yao, X. Huang, B. Zhang, Y. Li, J. Liu, Y. Wang and Z. Zhao, *Bioact. Mater.*, 2021, **6**, 4110–4140.
- 20 J. F. Liao, R. X. Han, Y. Z. Wu and Z. Y. Qian, *Bone Res.*, 2021, **9**(1), 18.
- 21 Q. Wang, J. Yan, J. Yang and B. Li, *Mater. Today*, 2016, **19**, 451–463.
- 22 T. Zhang, Q. Wei, H. Zhou, Z. Jing, X. Liu, Y. Zheng, H. Cai, F. Wei, L. Jiang, M. Yu, Y. Cheng, D. Fan, W. Zhou, X. Lin, H. Leng, J. Li, X. Li, C. Wang, Y. Tian and Z. Liu, *Bioact. Mater.*, 2021, **6**, 3659–3670.
- 23 W. J. Basirun, B. Nasiri-Tabrizi and S. Baradaran, *Crit. Rev. Solid State Mater. Sci.*, 2018, **43**, 177–212.
- 24 N. Wang, S. Thameem Dheen, J. Y. H. Fuh and A. Senthil Kumar, *Bioprinting*, 2021, **23**, e00146.
- 25 I. A. Urban, E. Montero, A. Monje and I. Sanz-Sanchez, *J. Clin. Periodontol.*, 2019, **46**, 319–339.
- 26 C. A. de Sousa, C. A. Araujo Lemos, J. F. Santiago-Junior, L. P. Faverani and E. P. Pellizzer, *J. Dent.*, 2018, **76**, 1–8.
- 27 A. Mostafavi, T. Abudula, C. S. Russell, E. Mostafavi, T. J. Williams, N. Salah, A. Alshahri, S. Harris, S. M. M. Basri, Y. K. Mishra, T. J. Webster, A. Memic and A. Tamayol, *Acta Biomater.*, 2021, **127**, 313–326.
- 28 D. Li, L. Deng, Z. Yang, X. Xie, P. Kang and Z. Tan, *J. Biomater. Appl.*, 2016, **30**, 1322–1333.
- 29 W. Wang and K. W. K. Yeung, *Bioact. Mater.*, 2017, **2**, 224–247.
- 30 Q. Zhang, K. Huang, J. Tan, X. Lei, L. Huang, Y. Song, Q. Li, C. Zou and H. Xie, *Chin. Chem. Lett.*, 2022, **33**, 1623–1626.
- 31 J. A. Lenis, P. Rico, J. L. G. Ribelles, M. A. Pacha-Olivenza, M. L. González-Martín and F. J. Bolívar, *Mater. Sci. Eng., C*, 2020, **116**, 111268.
- 32 Y. Chen, N. Kawazoe and G. Chen, *Acta Biomater.*, 2018, **67**, 341–353.



33 I. Erezuma, I. Lukin, M. Desimone, Y. S. Zhang, A. Dolatshahi-Pirouz and G. Orive, *Biomater. Adv.*, 2023, **146**, 213274.

34 T. Liu, X. Zhang, Y. Luo, Y. Huang and G. Wu, *Stem Cells Int.*, 2016, **2016**, 1416047.

35 J. Liu, L. Yang, K. Liu and F. Gao, *Front. Pharmacol.*, 2023, **14**, 1050954.

36 A. Shaabani and R. Sedghi, *Carbohydr. Polym.*, 2021, **264**, 118045.

37 B. W. Tan, Q. Tang, Y. J. Zhong, Y. L. Wei, L. F. He, Y. T. Wu, J. B. Wu and J. F. Liao, *Int. J. Oral Sci.*, 2021, **13**(1), 9.

38 Y. Yu, B. Sun, C. Yi and X. Mo, *Front. Mater. Sci.*, 2017, **11**, 93–105.

39 X. Su, T. Wang and S. Guo, *Regener. Ther.*, 2021, **16**, 63–72.

40 W.-C. Ji, X.-W. Zhang and Y.-S. Qiu, *World J. Exp. Med.*, 2016, **6**, 58–62.

41 J. R. Bush, H. Liang, M. Dickinson and E. A. Botchwey, *Polym. Adv. Technol.*, 2016, **27**, 1050–1055.

42 E. Quinlan, A. López-Noriega, E. Thompson, H. M. Kelly, S. A. Cryan and F. J. O'Brien, *J. Controlled Release*, 2015, **198**, 71–79.

43 J. L. Drury and D. J. Mooney, *Biomaterials*, 2003, **24**, 4337–4351.

44 Y. T. Wu, X. Zhang, Q. Zhao, B. W. Tan, X. Y. Chen and J. F. Liao, *J. Biomed. Nanotechnol.*, 2020, **16**, 1667–1686.

45 Y. Miao, Y. Chen, J. Luo, X. Liu, Q. Yang, X. Shi and Y. Wang, *Bioact. Mater.*, 2023, **21**, 97–109.

46 X. Lv, Y. Liu, S. Song, C. Tong, X. Shi, Y. Zhao, J. Zhang and M. Hou, *Carbohydr. Polym.*, 2019, **205**, 312–321.

47 R. Kadri, J. Bacharouch, K. Elkhoury, G. Ben Messaoud, C. Kahn, S. Desobry, M. Linder, A. Tamayol, G. Francius, J. F. Mano, L. Sanchez-Gonzalez and E. Arab-Tehrany, *Mater. Today Bio*, 2020, **6**, 100046.

48 I. Antoniuk, D. Kaczmarek, A. Kardos, I. Varga and C. Amiel, *Polymers*, 2018, **10**, 566.

49 S. R. Shin, C. Shin, A. Memic, S. Shadmehr, M. Miscuglio, H. Y. Jung, S. M. Jung, H. Bae, A. Khademhosseini, X. Tang and M. R. Dokmeci, *Adv. Funct. Mater.*, 2015, **25**, 4486–4495.

50 W. Bonani, N. Cagol and D. Maniglio, in *Biomimicked Biomaterials: Advances in Tissue Engineering and Regenerative Medicine*, ed. H. J. Chun, R. L. Reis, A. Motta and G. Khang, 2020, vol. 1250, pp. 49–61.

51 K. Elkhoury, C. S. Russell, L. Sanchez-Gonzalez, A. Mostafavi, T. J. Williams, C. Kahn, N. A. Peppas, E. Arab-Tehrany and A. Tamayol, *Adv. Healthcare Mater.*, 2019, **8**(18), 1900506.

52 C. Zhang, Q. Dong, K. Liang, D. Zhou, H. Yang, X. Liu, W. Xu, Y. Zhou and P. Xiao, *Int. J. Biol. Macromol.*, 2018, **119**, 270–277.

53 C. D. Spicer, *Polym. Chem.*, 2020, **11**, 184–219.

54 H. D. N. Tran, K. D. Park, Y. C. Ching, H. Cong and N. Dai Hai, *J. Ind. Eng. Chem.*, 2020, **89**, 58–82.

55 Q. Zhang, B. Shi, J. Ding, L. Yan, J. P. Thawani, C. Fu and X. Chen, *Acta Biomater.*, 2019, **88**, 57–77.

56 S. Sun, L.-B. Mao, Z. Lei, S.-H. Yu and H. Coelfen, *Angew. Chem., Int. Ed.*, 2016, **55**, 11765–11769.

57 Q. Zeng, M. S. Desai, H.-E. Jin, J. H. Lee, J. Chang and S.-W. Lee, *Biomacromolecules*, 2016, **17**, 2619–2625.

58 J. Qu, X. Zhao, Y. Liang, T. Zhang, P. X. Ma and B. Guo, *Biomaterials*, 2018, **183**, 185–199.

59 R. Narayanaswamy and V. P. Torchilin, *Molecules*, 2019, **24**(3), 603.

60 Z. Bao, P. Gao, G. Xia, Z. Wang, M. Kong, C. Feng, X. Cheng, Y. Liu and X. Chen, *J. Mater. Chem. B*, 2016, **4**, 3936–3944.

61 M. Grosjean, L. Gangolphe and B. Nottelet, *Adv. Funct. Mater.*, 2023, **33**, 2205315.

62 Y. Y. Tang, X. M. Sun, J. C. Ma and Q. S. Yan, *J. Biomater. Sci., Polym. Ed.*, 2023, 1–22.

63 H. Hamed, S. Moradi, S. M. Hudson and A. E. Tonelli, *Carbohydr. Polym.*, 2018, **199**, 445–460.

64 G. Xia, Y. Liu, M. Tian, P. Gao, Z. Bao, X. Bai, X. Yu, X. Lang, S. Hu and X. Chen, *J. Mater. Chem. B*, 2017, **5**, 3172–3185.

65 M. M. Qin, Y. Q. Guo, F. F. Su, X. P. Huang, Q. P. Qian, Y. L. Zhou and J. Y. Pan, *Chem. Eng. J.*, 2023, **455**, 140854.

66 X. Jing, H.-Y. Mi, Y.-J. Lin, E. Enriquez, X.-F. Peng and L.-S. Turng, *ACS Appl. Mater. Interfaces*, 2018, **10**, 20897–20909.

67 J. Tavakoli and Y. Tang, *Polymers*, 2017, **9**(8), 364.

68 S. G. Son, H. J. Park, S. M. Kim, S. J. Kim, M. S. Kil, J. M. Jeong, Y. E. Lee, Y. Eom, S. Y. Hwang, J. Park and B. G. Choi, *Chem. Eng. J.*, 2023, **454**, 140443.

69 Q. Ling, X. Fan, M. Ling, J. Liu, L. Zhao and H. Gu, *ACS Appl. Mater. Interfaces*, 2023, **15**, 12350–12362.

70 E. M. Ahmed, *J. Adv. Res.*, 2015, **6**, 105–121.

71 N. N. Dil and M. Sadeghi, *J. Hazard. Mater.*, 2018, **351**, 38–53.

72 C. Shao, M. Wang, L. Meng, H. Chang, B. Wang, F. Xu, J. Yang and P. Wan, *Chem. Mater.*, 2018, **30**, 3110–3121.

73 X. Liu, C. Tang, X. Du, S. Xiong, S. Xi, Y. Liu, X. Shen, Q. Zheng, Z. Wang, Y. Wu, A. Horner and J.-K. Kim, *Mater. Horiz.*, 2017, **4**, 477–486.

74 Z. Lei, Q. Wang, S. Sun, W. Zhu and P. Wu, *Adv. Mater.*, 2017, **29**(22), DOI: [10.1002/adma.201700321](https://doi.org/10.1002/adma.201700321).

75 C. Pascual-Garrido, F. Rodriguez-Fontan, E. A. Aisenbrey, K. A. Payne, J. Chahla, L. R. Goodrich and S. J. Bryant, *J. Orthop. Res.*, 2018, **36**, 64–75.

76 M. Mehrali, A. Thakur, C. P. Pennisi, S. Talebian, A. Arpanaei, M. Nikkhah and A. Dolatshahi-Pirouz, *Adv. Mater.*, 2017, **29**(8), DOI: [10.1002/adma.201603612](https://doi.org/10.1002/adma.201603612).

77 I. M. Tayler and R. S. Stowers, *Acta Biomater.*, 2021, **132**, 4–22.

78 H. T. Liao, M.-J. Tsai, M. Brahmaya and J.-P. Chen, *Int. J. Mol. Sci.*, 2018, **19**(9), 2537.

79 C. Acevedo, V. A. Stadelmann, D. P. Pioletti, T. Alliston and R. O. Ritchie, *Nat. Biomed. Eng.*, 2018, **2**, 62–71.

80 L. Shi, P. Ding, Y. Wang, Y. Zhang, D. Ossipov and J. Hilborn, *Macromol. Rapid Commun.*, 2019, **40**(7), 1800837.

81 W. Zhu, J. Zhang, Z. Wei, B. Zhang and X. Weng, *Materials*, 2023, **16**(3), 1215.

82 M. Diba, S. Spaans, K. Ning, B. D. Ippel, F. Yang, B. Loomans, P. Y. W. Dankers and S. C. G. Leeuwenburgh, *Adv. Mater. Interfaces*, 2018, **5**(17), 1800118.



83 J. F. Patrick, M. J. Robb, N. R. Sottos, J. S. Moore and S. R. White, *Nature*, 2016, **540**, 363–370.

84 Z. B. Liu, J. Li, Z. P. Zhang, J. Z. Liu, C. Y. Wu and Y. Q. Yu, *Eur. Polym. J.*, 2023, **182**, 111728.

85 Y. L. Fang, X. S. Du, Z. L. Du, H. B. Wang and X. Cheng, *J. Mater. Chem. A*, 2017, **5**, 8010–8017.

86 Y. Wang, S. Zhang and J. Wang, *Chin. Chem. Lett.*, 2021, **32**, 1603–1614.

87 A. Sun, X. He, X. Ji, D. Hu, M. Pan, L. Zhang and Z. Qian, *Chin. Chem. Lett.*, 2021, **32**, 2117–2126.

88 A. Sreedevi Madhavikutty, A. K. Singh Chandel, C.-C. Tsai, N. F. Inagaki, S. Ohta and T. Ito, *Sci. Technol. Adv. Mater.*, 2023, **24**, 2175586.

89 W. Zhu, J. Y. Zhang, Z. Q. Wei, B. Z. Zhang and X. S. Weng, *Materials*, 2023, **16**(3), 1215.

90 L. Zhang, Z. Liu, X. Wu, Q. Guan, S. Chen, L. Sun, Y. Guo, S. Wang, J. Song, E. M. Jeffries, C. He, F.-L. Qing, X. Bao and Z. You, *Adv. Mater.*, 2019, **31**(23), e1901402.

91 S. Bai, M. Zhang, X. Huang, X. Zhang, C. Lu, J. Song and H. Yang, *Chem. Eng. J.*, 2021, **413**, 127512.

92 Y. Y. Yang, L. F. Xu, J. F. Wang, Q. Y. Meng, S. L. Zhong, Y. Gao and X. J. Cui, *Carbohydr. Polym.*, 2022, **283**, 119161.

93 S. Yue, H. He, B. Li and T. Hou, *Nanomaterials*, 2020, **10**, 1511.

94 X. Lin, M. Wang, J. Zhao, X. Wu, J. Xie and J. Yang, *Carbohydr. Polym.*, 2023, **304**, 120502.

95 C. E. Diesendruck, N. R. Sottos, J. S. Moore and S. R. White, *Angew. Chem., Int. Ed.*, 2015, **54**, 10428–10447.

96 C. Bao, Y.-J. Jiang, H. Zhang, X. Lu and J. Sun, *Adv. Funct. Mater.*, 2018, **28**(23), 1800560.

97 M. F. Zhang, X. Chen, K. D. Yang, Q. Dong, H. J. Yang, S. J. Gu, W. L. Xu and Y. S. Zhou, *Carbohydr. Polym.*, 2023, **301**, 120372.

98 M. Verhulsel, M. Vignes, S. Descroix, L. Malaquin, D. M. Vignjevic and J.-L. Viovy, *Biomaterials*, 2014, **35**, 1816–1832.

99 N. Annabi, J. W. Nichol, X. Zhong, C. Ji, S. Koshy, A. Khademhosseini and F. Dehghani, *Tissue Eng., Part B*, 2010, **16**, 371–383.

100 S. H. Oh, I. K. Park, J. M. Kim and J. H. Lee, *Biomaterials*, 2007, **28**, 1664–1671.

101 Z. Lan, R. Kar, M. Chwatko, E. Shoga and E. Cosgriff-Hernandez, *J. Biomed. Mater. Res., Part A*, 2023, **111**, 465–477.

102 M. Bercea, *Polymers*, 2022, **14**(12), 2365.

103 H. Cao, L. Duan, Y. Zhang, J. Cao and K. Zhang, *Signal Transduction Targeted Ther.*, 2021, **6**, 426.

104 S. Gerecht, S. A. Townsend, H. Pressler, H. Zhu, C. L. E. Nijst, J. P. Bruggeman, J. W. Nichol and R. Langer, *Biomaterials*, 2007, **28**, 4826–4835.

105 M. Bercea, *Polymers*, 2022, **14**(12), 2365.

106 Y. Xiao, E. A. Friis, S. H. Gehrke and M. S. Detamore, *Tissue Eng., Part B*, 2013, **19**, 403–412.

107 A. Vedadghavami, F. Minooei, M. H. Mohammadi, S. Khetani, A. Rezaei Kolahchi, S. Mashayekhan and A. Sanati-Nezhad, *Acta Biomater.*, 2017, **62**, 42–63.

108 S. Naahidi, M. Jafari, M. Logan, Y. Wang, Y. Yuan, H. Bae, B. Dixon and P. Chen, *Biotechnol. Adv.*, 2017, **35**, 530–544.

109 E. Caló and V. V. Khutoryanskiy, *Eur. Polym. J.*, 2015, **65**, 252–267.

110 K. Areevijit, N. Dhanesuan, J. A. Luckanagul and S. Rungsianont, *J. Biomater. Appl.*, 2020, **35**, 1294–1303.

111 J. Cao, G. He, X. Ning, X. Chen, L. Fan, M. Yang, Y. Yin and W. Cai, *Carbohydr. Polym.*, 2022, **287**, 119318.

112 E. Alsberg, H. J. Kong, Y. Hirano, M. K. Smith, A. Albeiruti and D. J. Mooney, *J. Dent. Res.*, 2003, **82**, 903–908.

113 Q. Huang, Y. Zou, M. C. Arno, S. Chen, T. Wang, J. Gao, A. P. Dove and J. Du, *Chem. Soc. Rev.*, 2017, **46**, 6255–6275.

114 A. Raza and C. C. Lin, *Macromol. Biosci.*, 2013, **13**, 1048–1058.

115 S. Khetan, M. Guvendiren, W. R. Legant, D. M. Cohen, C. S. Chen and J. A. Burdick, *Nat. Mater.*, 2013, **12**, 458–465.

116 X. Zhang and R. M. Waymouth, *J. Am. Chem. Soc.*, 2017, **139**, 3822–3833.

117 S. Gupta, A. Sharma, J. V. Kumar, V. Sharma, P. K. Gupta and R. S. Verma, *Int. J. Biol. Macromol.*, 2020, **162**, 1358–1371.

118 X. Yang, H. Yang, X. Jiang, B. Yang, K. Zhu, N. C.-H. Lai, C. Huang, C. Chang, L. Bian and L. Zhang, *Carbohydr. Polym.*, 2021, **256**, 117574.

119 S. Chen, F. Tang, L. Tang and L. Li, *ACS Appl. Mater. Interfaces*, 2017, **9**, 20895–20903.

120 J. L. Ruben, F. J. M. Roeters, A. F. Montagner and M. C. D. N. J. M. Huysmans, *J. Mech. Behav. Biomed. Mater.*, 2014, **30**, 75–82.

121 H. Wang, M. Bongio, K. Farbod, A. W. G. Nijhuis, J. van den Beucken, O. C. Boerman, J. C. M. van Hest, Y. Li, J. A. Jansen and S. C. G. Leeuwenburgh, *Acta Biomater.*, 2014, **10**, 508–519.

122 X. Zhang, B. Tan, Y. Wu, M. Zhang, X. Xie and J. Liao, *Carbohydr. Polym.*, 2022, **293**, 119722.

123 L. Cai, S. Liu, J. Guo and Y.-G. Jia, *Acta Biomater.*, 2020, **113**, 84–100.

124 W. Zhang, R. Wang, Z. Sun, X. Zhu, Q. Zhao, T. Zhang, A. Cholewinski, F. Yang, B. Zhao, R. Pinnaratip, P. K. Forooshani and B. P. Lee, *Chem. Soc. Rev.*, 2020, **49**, 433–464.

125 L. Shi, F. Wang, W. Zhu, Z. Xu, S. Fuchs, J. Hilborn, L. Zhu, Q. Ma, Y. Wang, X. Weng and D. A. Ossipov, *Adv. Funct. Mater.*, 2017, **27**(37), 1700591.

126 J. C. Cremaldi and B. Bhushan, *Beilstein J. Nanotechnol.*, 2018, **9**, 907–935.

127 J. Li, W. Li, M. Kong, Z. Li, T. Yang, Q. Wang and W. Teng, *J. Nanobiotechnol.*, 2023, **21**, 62.

128 J. Qu, X. Zhao, P. X. Ma and B. Guo, *Acta Biomater.*, 2017, **58**, 168–180.

129 Z. Wang, Y. Ren, Y. Zhu, L. Hao, Y. Chen, G. An, H. Wu, X. Shi and C. Mao, *Angew. Chem., Int. Ed.*, 2018, **57**, 9008–9012.

130 N. Pathan and P. Shende, *Mater. Sci. Eng., C*, 2021, **125**, 112099.

131 S. Y. An, D. Arunbabu, S. M. Noh, Y. K. Song and J. K. Oh, *Chem. Commun.*, 2015, **51**, 13058–13070.



132 L. Yue, X. Zhang, Y. Wang, W. Li, Y. Tang and Y. Bai, *Eur. Polym. J.*, 2021, **146**, 110258.

133 M. I. Shekh, G. M. Zhu, W. Xiong, W. L. Wu, F. J. Stadler, D. Patel and C. T. Zhu, *Int. J. Biol. Macromol.*, 2023, **224**, 604–620.

134 M. F. Akhtar, M. Hanif and N. M. Ranjha, *Saudi Pharm. J.*, 2016, **24**, 554–559.

135 J. Y. C. Lim, Q. Lin, K. Xue and X. J. Loh, *Mater. Today Adv.*, 2019, **3**, 100021.

136 B. D. Rabideau and A. E. Ismail, *Phys. Chem. Chem. Phys.*, 2015, **17**, 5767–5775.

137 Y. Wang, *Biomaterials*, 2018, **178**, 663–680.

138 W. Wang, Y. Zhang and W. Liu, *Prog. Polym. Sci.*, 2017, **71**, 1–25.

139 J. Chen, X. Xu, M. Liu, Y. Li, D. Yu, Y. Lu, M. Xiong, I. Wyman, X. Xu and X. Wu, *Carbohydr. Polym.*, 2021, **264**, 117978.

140 D. Wang, M. Wagner, H.-J. Butt and S. Wu, *Soft Matter*, 2015, **11**, 7656–7662.

141 K. Miyamae, M. Nakahata, Y. Takashima and A. Harada, *Angew. Chem., Int. Ed.*, 2015, **54**, 8984–8987.

142 Q. Feng, K. Wei, S. Lin, Z. Xu, Y. Sun, P. Shi, G. Li and L. Bian, *Biomaterials*, 2017, **112**, 346–347.

143 N. Ding, X. Cai, P. Zhang, S. Dong, B. Du, J. Nie and P. Yu, *ACS Appl. Polym. Mater.*, 2021, **3**, 2709–2721.

144 J. Ye, S. Fu, S. Zhou, M. Li, K. Li, W. Sun and Y. Zhai, *Eur. Polym. J.*, 2020, **139**, 110024.

145 M. Chen, X. Ren, L. Dong, X. Li and H. Cheng, *Int. J. Biol. Macromol.*, 2021, **182**, 1259–1267.

146 C.-H. Li and J.-L. Zuo, *Adv. Mater.*, 2020, **32**, 1903762.

147 A. Alam, Q. Meng, G. Shi, S. Arabi, J. Ma, N. Zhao and H.-C. Kuan, *Compos. Sci. Technol.*, 2016, **127**, 119–126.

148 Y. Wu, X. Zhang, B. Tan, Y. Shan, X. Zhao and J. Liao, *Biomater. Adv.*, 2022, **133**, 112641.

149 D. Mawad, A. Artzy-Schnirman, J. Tonkin, J. Ramos, S. Inal, M. M. Mahat, N. Darwish, L. Zwi-Dantsis, G. G. Malliaras, J. J. Gooding, A. Lauto and M. M. Stevens, *Chem. Mater.*, 2016, **28**, 6080–6088.

150 Y. Xu, Y. Li, Q. Chen, L. Fu, L. Tao and Y. Wei, *Int. J. Mol. Sci.*, 2018, **19**(8), 2198.

151 Y. Tu, N. Chen, C. Li, H. Liu, R. Zhu, S. Chen, Q. Xiao, J. Liu, S. Ramakrishna and L. He, *Acta Biomater.*, 2019, **90**, 1–20.

152 B. Chen, Y. Liang, J. Zhang, L. Bai, M. Xu, Q. Han, X. Han, J. Xiu, M. Li, X. Zhou, B. Guo and Z. Yin, *Theranostics*, 2021, **11**, 5911–5925.

153 M. Wu, J. Chen, W. Huang, B. Yan, Q. Peng, J. Liu, L. Chen and H. Zeng, *Biomacromolecules*, 2020, **21**, 2409–2420.

154 H. L. Hou, L. Cardo, D. Mancino, B. Arnaiz, A. Criado and M. Prato, *Nanoscale*, 2020, **12**, 23824–23830.

155 J. Lei, X. Li, S. Wang, L. Yuan, L. Ge, D. Li and C. Mu, *ACS Appl. Polym. Mater.*, 2019, **1**, 1350–1358.

156 S.-W. Wu, X. Liu, A. L. Miller II, Y.-S. Cheng, M.-L. Yeh and L. Lu, *Carbohydr. Polym.*, 2018, **192**, 308–316.

157 V. T. Tran, M. T. I. Mredha, J. Y. Na, J.-K. Seon, J. Cui and I. Jeon, *Chem. Eng. J.*, 2020, **394**, 124941.

158 Y. Cao, Y. Yao, Y. Li, X. Yang, Z. Cao and G. Yang, *J. Colloid Interface Sci.*, 2019, **544**, 121–129.

159 A. Skardal, J. Zhang, L. McCoard, S. Oottamasathien and G. D. Prestwich, *Adv. Mater.*, 2010, **22**, 4736–4740.

160 H. An, Y. Bo, D. Chen, Y. Wang, H. Wang, Y. He and J. Qin, *RSC Adv.*, 2020, **10**, 11300–11310.

161 W. Du, A. Deng, J. Guo, J. Chen, H. Li and Y. Gao, *Carbohydr. Polym.*, 2019, **223**, 115084.

162 T. Kajisa and T. Sakata, *Sci. Technol. Adv. Mater.*, 2017, **18**, 26–33.

163 M. Montiel-Herrera, A. Gandini, F. M. Goycoolea, N. E. Jacobsen, J. Lizardi-Mendoza, M. Recillas-Mota and W. M. Argüelles-Monal, *Carbohydr. Polym.*, 2015, **128**, 220–227.

164 F. Yu, X. Cao, L. Zeng, Q. Zhang and X. Chen, *Carbohydr. Polym.*, 2013, **97**, 188–195.

165 O. Guaresti, C. García-Astrain, R. H. Aguirresarobe, A. Eceiza and N. Gabilondo, *Carbohydr. Polym.*, 2018, **183**, 278–286.

166 G. R. Bardajee, F. Mizani and S. S. Hosseini, *J. Polym. Res.*, 2017, **24**(3), 48.

167 Y. Deng, I. Hussain, M. Kang, K. Li, F. Yao, S. Liu and G. Fu, *Chem. Eng. J.*, 2018, **353**, 900–910.

168 T. Takei, R. Yoshihara, S. Danjo, Y. Fukuhara, C. Evans, R. Tomimatsu, Y. Ohzuno and M. Yoshida, *Int. J. Biol. Macromol.*, 2020, **149**, 140–147.

169 H. Jiang, L. Duan, X. Ren and G. Gao, *Eur. Polym. J.*, 2019, **112**, 660–669.

170 S. M. Mantooth, B. G. Munoz-Robles and M. J. Webber, *Macromol. Biosci.*, 2019, **19**, 1800281.

171 R. Yu, Y. Yang, J. He, M. Li and B. Guo, *Chem. Eng. J.*, 2021, **417**, 128278.

172 H.-H. Lee, Y.-W. Kim, J. Woo, H. Park, K. Hur, Z. Suo and J.-Y. Sun, *Extreme Mechanics Letters*, 2019, **33**, 100572.

173 T. Kopač, M. Krajnc and A. Ručigaj, *Int. J. Biol. Macromol.*, 2021, **168**, 695–707.

174 A. Wang, Y. Wang, B. Zhang, K. Wan, J. Zhu, J. Xu, C. Zhang and T. Liu, *Chem. Eng. J.*, 2021, **411**, 128506.

175 J. Cao, Y. Wang, C. He, Y. Kang and J. Zhou, *Carbohydr. Polym.*, 2020, **242**, 116420.

176 W. Tanaka, H. Shigemitsu, T. Fujisaku, R. Kubota, S. Minami, K. Urayama and I. Hamachi, *J. Am. Chem. Soc.*, 2019, **141**, 4997–5004.

177 V. K. Anupama Devi, R. Shyam, A. Palaniappan, A. K. Jaiswal, T.-H. Oh and A. J. Nathanael, *Polymers*, 2021, **13**(21), 3782.

178 S. Talebian, M. Mehrali, N. Taebnia, C. P. Pennisi, F. B. Kadumudi, J. Foroughi, M. Hasany, M. Nikkhah, M. Akbari, G. Orive and A. Dolatshahi-Pirouz, *Adv. Sci.*, 2019, **6**, 1801664.

179 N. Lagneau, L. Terriac, P. Tournier, J. J. Helesbeux, G. Viault, D. Séraphin, B. Halgand, F. Loll, C. Garnier, C. Jonchère, M. Rivière, A. Tessier, J. Lebreton, Y. Maugars, J. Guicheux, C. Le Visage and V. Delplace, *Biomater. Sci.*, 2023, **11**, 2033–2045.

180 C. C. Deng, W. L. A. Brooks, K. A. Abboud and B. S. Sumerlin, *ACS Macro Lett.*, 2015, **4**, 220–224.

181 D.-q. Li, S.-y. Wang, Y.-j. Meng, Z.-w. Guo, M.-m. Cheng and J. Li, *Carbohydr. Polym.*, 2021, **268**, 118244.



182 A. E. G. Baker, R. Y. Tam and M. S. Shoichet, *Biomacromolecules*, 2017, **18**, 4373–4384.

183 L. Ma, W. Su, Y. Ran, X. Ma, Z. Yi, G. Chen, X. Chen, Z. Deng, Q. Tong, X. Wang and X. Li, *Int. J. Biol. Macromol.*, 2020, **165**, 1164–1174.

184 P. Pan, X. Chen, H. Xing, Y. Deng, J. Chen, F. A. Alharthi, A. A. Alghamdi and J. Su, *Chin. Chem. Lett.*, 2021, **32**, 2159–2163.

185 X. Bai, S. Lü, Z. Cao, B. Ni, X. Wang, P. Ning, D. Ma, H. Wei and M. Liu, *Carbohydr. Polym.*, 2017, **166**, 123–130.

186 S. Lu, X. Bai, H. Liu, P. Ning, Z. Wang, C. Gao, B. Ni and M. Liu, *J. Mater. Chem. B*, 2017, **5**, 3739–3748.

187 S. Bi, J. Pang, L. Huang, M. Sun, X. Cheng and X. Chen, *Int. J. Biol. Macromol.*, 2020, **146**, 99–109.

188 Z. Xu, D.-G. Hwang, M. D. Bartlett, S. Jiang and K. M. Bratlie, *Biochem. Eng. J.*, 2021, **165**, 107821.

189 Q. Feng, K. Wei, S. Lin, Z. Xu, Y. Sun, P. Shi, G. Li and L. Bian, *Biomaterials*, 2016, **101**, 217–228.

190 S. R. Caliari and J. A. Burdick, *Nat. Methods*, 2016, **13**, 405–414.

191 A. M. Rosales and K. S. Anseth, *Nat. Rev. Mater.*, 2016, **1**(2), 15012.

192 X. Yang, G. Liu, L. Peng, J. Guo, L. Tao, J. Yuan, C. Chang, Y. Wei and L. Zhang, *Adv. Funct. Mater.*, 2017, **27**(40), 1703174.

193 M. H. Turabee, T. Thambi, D. Huu Thuy Trang, J. H. Jeong and D. S. Lee, *Biomater. Sci.*, 2018, **6**, 661–671.

194 L. Bian, C. Hou, E. Tous, R. Rai, R. L. Mauck and J. A. Burdick, *Biomaterials*, 2013, **34**, 413–421.

195 Q. Feng, J. Xu, K. Zhang, H. Yao, N. Zheng, L. Zheng, J. Wang, K. Wei, X. Xiao, L. Qin and L. Bian, *ACS Cent. Sci.*, 2019, **5**, 440–450.

196 C. I. A. van Houdt, D. A. Cardoso, B. A. J. A. van Oirschot, D. J. O. Ulrich, J. A. Jansen, S. C. G. Leeuwenburgh and J. J. J. P. van den Beucken, *J. Tissue Eng. Regener. Med.*, 2017, **11**, 2537–2548.

197 Z. W. Deng, T. B. Qian and F. Hang, *ACS Biomater. Sci. Eng.*, 2020, **6**, 7061–7070.

198 K. Huang, J. Huang, J. Zhao, Z. Gu and J. Wu, *Chin. Chem. Lett.*, 2022, **33**, 1941–1945.

199 H. Zhang, Y. Cong, A. R. Osi, Y. Zhou, F. Huang, R. P. Zaccaria, J. Chen, R. Wang and J. Fu, *Adv. Funct. Mater.*, 2020, **30**(13), 1910573.

200 Y. Zhao, C. Gao, H. Liu, H. Liu, Y. Feng, Z. Li, H. Liu, J. Wang, B. Yang and Q. Lin, *Acta Biomater.*, 2021, **121**, 653–664.

201 L. Zhao, X. Zhang, X. Wang, X. Guan, W. Zhang and J. Ma, *J. Nanobiotechnol.*, 2021, **19**, 335.

202 Z. Wan, P. Zhang, L. Lv and Y. Zhou, *Theranostics*, 2020, **10**, 11837–11861.

203 J. Liao, K. Shi, Y. Jia, Y. Wu and Z. Qian, *Bioact. Mater.*, 2021, **6**, 2221–2230.

204 Y. Wu, X. Zhang, B. Tan, Y. Shan, X. Zhao and J. Liao, *Biomater. Adv.*, 2022, **133**, 112641.

205 S. Luo, J. Wu, Z. Jia, P. Tang, J. Sheng, C. Xie, C. Liu, D. Gan, D. Hu, W. Zheng and X. Lu, *Macromol. Biosci.*, 2019, **19**, e1900047.

206 N. Xu, X. Zhang, T. Qi, Y. Wu, X. Xie, F. Chen, D. Shao and J. Liao, *MedComm*, 2022, **1**, e25.

207 K. E. Matheny, E. Y. Tseng, K. B. Carter Jr, W. B. Cobb and K. J. Fong, *American Journal of Rhinology & Allergy*, 2014, **28**, 508–513.

208 A. Machado, I. Pereira, F. Costa, A. Brandão, J. E. Pereira, A. C. Maurício, J. D. Santos, I. Amaro, R. Falacho, R. Coelho, N. Cruz and M. Gama, *Clin. Oral Investig.*, 2023, **27**, 979–994.

209 K. Flegeau, R. Pace, H. Gautier, G. Rethore, J. Guicheux, C. Le Visage and P. Weiss, *Adv. Colloid Interface Sci.*, 2017, **247**, 589–609.

210 G. Chang, Y. Chen, Y. Li, S. Li, F. Huang, Y. Shen and A. Xie, *Carbohydr. Polym.*, 2015, **122**, 336–342.

211 A. Zhang, Y. Liu, D. Qin, M. Sun, T. Wang and X. Chen, *Int. J. Biol. Macromol.*, 2020, **164**, 2108–2123.

212 D. J. Quick and K. S. Anseth, *J. Controlled Release*, 2004, **96**, 341–351.

213 D. Costa, A. J. Valente, M. G. Miguel and J. Queiroz, *Adv. Colloid Interface Sci.*, 2014, **205**, 257–264.

214 H. Storrie and D. J. Mooney, *Adv. Drug Delivery Rev.*, 2006, **58**, 500–514.

215 M. Dadsetan, J. P. Szatkowski, K. L. Shogren, M. J. Yaszemski and A. Maran, *J. Biomed. Mater. Res., Part A*, 2009, **91**, 1170–1177.

216 K. Komatsu, T. Shibata, A. Shimada, H. Ideno, K. Nakashima, Y. Tabata and A. Nifuji, *J. Biomater. Sci., Polym. Ed.*, 2016, **27**, 419–430.

217 M. D. Krebs, E. Salter, E. Chen, K. A. Sutter and E. Alsberg, *J. Biomed. Mater. Res., Part A*, 2010, **92**, 1131–1138.

218 H. J. Haugen, S. P. Lyngstadaas, F. Rossi and G. Perale, *J. Clin. Periodontol.*, 2019, **46**, 92–102.

219 R. J. Miron, Q. Zhang, A. Sculean, D. Buser, B. E. Pippenger, M. Dard, Y. Shirakata, F. Chandad and Y. Zhang, *Clin. Oral Investig.*, 2016, **20**, 2259–2265.

220 A. Mandal, J. R. Clegg, A. C. Anselmo and S. Mitragotri, *Bioeng. Transl. Med.*, 2020, **5**, e10158.

221 A. J. Vegas, O. Veiseh, J. C. Doloff, M. Ma, H. H. Tam, K. Bratlie, J. Li, A. R. Bader, E. Langan, K. Olejnik, P. Fenton, J. W. Kang, J. Hollister-Locke, M. A. Bochenek, A. Chiu, S. Siebert, K. Tang, S. Jhunjhunwala, S. Aresta-Dasilva, N. Dholakia, R. Thakrar, T. Vietti, M. Chen, J. Cohen, K. Siniakowicz, M. Qi, J. McGarrigle, A. C. Graham, S. Lyle, D. M. Harlan, D. L. Greiner, J. Oberholzer, G. C. Weir, R. Langer and D. G. Anderson, *Nat. Biotechnol.*, 2016, **34**, 345–352.

222 S. Talebian, M. Mehrali, N. Taebnia, C. P. Pennisi, F. B. Kadumudi, J. Foroughi, M. Hasany, M. Nikkhah, M. Akbari, G. Orive and A. Dolatshahi-Pirouz, *Adv. Sci.*, 2019, **6**(16), 1801664.

223 A. Rafique, K. Mahmood Zia, M. Zuber, S. Tabasum and S. Rehman, *Int. J. Biol. Macromol.*, 2016, **87**, 141–154.

224 M. Zare, A. Bigham, M. Zare, H. Luo, E. Rezvani Ghomi and S. Ramakrishna, *Int. J. Mol. Sci.*, 2021, **22**(12), 6376.

225 S. Cascone and G. Lamberti, *Int. J. Pharm.*, 2020, **573**, 118803.

