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# Unveiling the unprecedented catalytic capability of micro-sized Co-ZIF-L for the thermal decomposition of RDX by 2D-structure-induced mechanism reversal†

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Developing MOF-based catalysts with superior catalytic properties for the thermal decomposition of cyclotrimethylenetrinitramine (RDX) is significant for the application of novel and efficient combustion catalysts oriented to RDX-based propellants with excellent combustion performance. Herein, micro-sized Co-ZIF-L with a star-like morphology (SL-Co-ZIF-L) was found to exhibit unprecedented catalytic capability for the decomposition of RDX, which can lower the decomposition temperature of RDX by 42.9 °C and boost the heat release by 50.8%, superior to that of all the ever-reported MOFs and even ZIF-67, which has similar chemical composition but a much smaller size. In-depth mechanism study from both experimental and theoretical views reveals that the weekly interacted 2D layered structure of SL-Co-ZIF-L could activate the exothermic C–N fission pathway for the decomposition of RDX in the condensed phase, thus reversing the commonly advantageous N–N fission pathway and promoting the decomposition process in the low-temperature stage. Our study reveals the unusually superior catalytic capability of micro-sized MOF catalysts and sheds light on the rational structure design of catalysts used in micromolecule transformation reactions, typically the thermal decomposition of energetic materials.

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## Introduction

As a significant sort of launching energy, propellants have the ability to produce high-temperature gases through rapid and regular combustion reactions, so as to propel projectiles or rockets. Thereinto, the combustion performance of propellants determines the rate and regularity of energy release, which plays a crucial role for the power of weapon platforms. Adding combustion catalysts into propellants can effectively modulate their thermodecomposition rate and further boost the combustion performance, thus finally enhancing the energy-releasing efficiency.<sup>1–3</sup>

Nowadays, the widely used combustion catalysts mainly include metal oxides,<sup>4–6</sup> metal salts,<sup>7,8</sup> metal complexes,<sup>9,10</sup> carbon materials<sup>11,12</sup> and various composite catalysts.<sup>13–16</sup> Compared with the above-mentioned catalysts, metal–organic frameworks (MOFs) have unique advantages in catalytic combustion reactions, thus becoming an increasingly hot

material as a combustion catalyst in recent years.<sup>17–20</sup> Firstly, MOFs (especially stable MOFs<sup>21,22</sup>) possess an ordered and controllable porous structure, which can help to realize selective adsorption and reaction of different reactants and intermediates. Secondly, the highly dispersed metal centres in MOFs can not only enhance the contact between active sites and propellant components, which boosts the atom utilization of catalysts, but also inhibit the agglomeration and inactivation of active centres. Furthermore, the *in situ* formed metal–carbon or metal–oxide–carbon composite materials during the catalytic combustion reactions are also efficient catalysts with high stability.<sup>23,24</sup> These characteristics make MOFs a type of novel and high-performance combustion catalyst.

However, probably due to the lack of elaborate regulations towards the composition and structure of catalysts, as well as in-depth excavation of the mechanisms and structure–activity relationships, MOF-based combustion catalysts have not exhibited superior catalytic performance compared with traditional catalysts. Besides, existing MOF catalysts are mainly aimed at the catalytic combustion process of ammonium perchlorate (AP), one of the most commonly used oxidizer in solid propellants.<sup>17,19–23,25–27</sup> Nevertheless, there have been few research studies targeted on the catalytic effect of MOFs towards the thermal decomposition of cyclotrimethylenetrinitramine (RDX), another high-energy additive, which is widely used in

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† Electronic supplementary information (ESI) available: Experimental section, SEM images, FT-IR spectra, XPS spectra, TG-DTG curves, DSC curves, MS curves, crystal structure, micropore size distribution curve, free energy for different decomposition pathways and Tables S1 and S2. See DOI: <https://doi.org/10.1039/d3ra01551c>



both solid propellants and gun propellants. As a consequence of the above two points, MOF-based combustion catalysts with superior catalytic performance for RDX have not been reported, even if some energetic MOFs (EMOFs) have also been synthesized and tried as combustion catalysts.<sup>28–34</sup>

Based on the above-mentioned situation, we herein took advantage of star-like Co-ZIF-L (abbreviated as SL-Co-ZIF-L), which was synthesized *via* a room-temperature reaction using 2-methylimidazole (2-MI) as the ligand and H<sub>2</sub>O as the only solvent, as an efficient catalyst for the thermal decomposition of RDX. Unexpectedly, this micro-sized catalyst shows an unprecedented catalytic capability compared with traditional catalysts and existing MOF catalysts, which can lower the decomposition temperature of RDX by 42.9 °C and boost the heat release by 50.8%. As far as our knowledge, such catalytic performance is far superior to that of all the ever-reported MOF catalysts as well as ZIF-67, which has the same ligand but a different crystal structure compared with Co-ZIF-L, under similar test conditions. More importantly, the inherent reasons for the rather high catalytic property of SL-Co-ZIF-L were thoroughly investigated by multiple methods. The results show that although the morphology and structure of SL-Co-ZIF-L have evident disadvantages in the enlargement of specific surface area, the decomposition of RDX can be accelerated through a distinctive reaction mechanism, where the exothermic C–N fission pathway can be activated for the decomposition of RDX in the condensed phase due to the weekly interacted 2D layered structure of SL-Co-ZIF-L, thus reversing the advantageous N–N fission pathway in common cases. As a consequence, SL-Co-ZIF-L will even promote the decomposition of RDX before melting and enhance the heat release during the whole decomposition process, which is unprecedented among both energetic and non-energetic combustion catalysts.

## Results and discussion

### Synthesis and characterization of SL-Co-ZIF-L

SL-Co-ZIF-L was first synthesized by a straightforward aqueous-phase reaction at room temperature. Similar to ZIF-67, which is widely used as a MOF-based catalyst or catalyst precursor in the areas of thermocatalysis,<sup>35,36</sup> photocatalysis<sup>37,38</sup> and electrocatalysis,<sup>39–41</sup> Co(NO<sub>3</sub>)<sub>2</sub>·6H<sub>2</sub>O and 2-MI were chosen as the metal source and organic ligand, respectively. Nevertheless, deionized water was selected to substitute methanol for the dissolution of the two reactants. After standing still for 18 h, the pinkish-purple product was precipitated in abundance from the solution containing Co(NO<sub>3</sub>)<sub>2</sub>·6H<sub>2</sub>O and 2-MI.

The morphology of the product was characterized by SEM and TEM. As revealed in Fig. 1a and b, SL-Co-ZIF-L shows a star-like hierarchical structure, which is made up of micro-rods with a length of approximately 3–7 μm and a width of around 1 μm. It is worth mentioning that the microrods seem to be angular rather than smooth in the side and top, seen from the magnified SEM and TEM images (Fig. 1c and d). Moreover, a similar element distribution of cobalt, carbon and nitrogen (Fig. 1e) indicates that the SL-Co-ZIF-L we obtained is

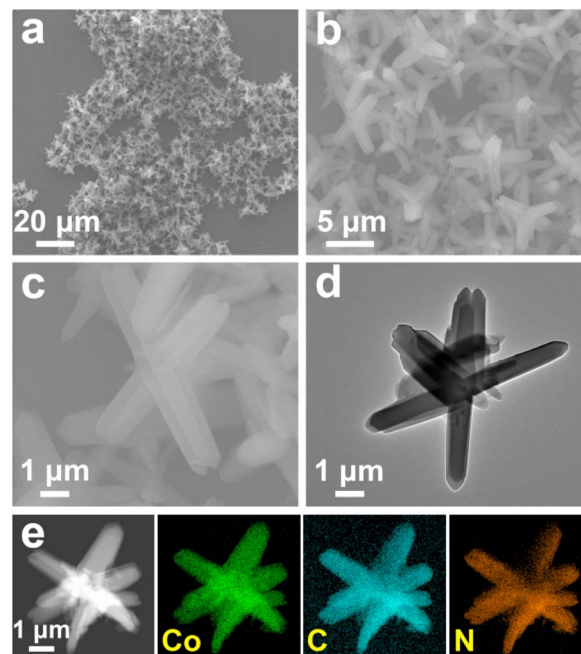


Fig. 1 (a–c) SEM, (d) TEM and (e) HAADF-STEM and the corresponding EDS mapping images of SL-Co-ZIF-L.

a nitrogen-containing Co-MOF instead of metal oxides or other materials.

XRD measurement was conducted to obtain the crystal structure information of Co-ZIF-L. From Fig. 2, the XRD pattern of SL-Co-ZIF-L is well in accordance with the simulated orthorhombic Co-ZIF-L rather than ZIF-67,<sup>42</sup> and no obvious impurities can be observed. It is necessary to point out that Co-ZIF-L and ZIF-67 are two Co-MOFs with the same ligand but different topological structures, just like the relationship between Zn-ZIF-L and ZIF-8.<sup>42–44</sup> ZIF-67 with a rhombic dodecahedron morphology was also prepared for comparison, whose morphology and XRD pattern are shown in Fig. S1† and 2, respectively.

Infrared spectroscopic study was carried out to investigate the chemical compositions of SL-Co-ZIF-L. From Fig. S2,† the FT-IR spectrum of SL-Co-ZIF-L is highly consistent with that of the previously reported Co-ZIF-L.<sup>44</sup> It reveals similar

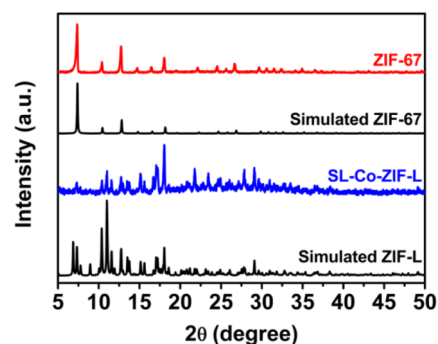


Fig. 2 XRD patterns of SL-Co-ZIF-L and ZIF-67.



characteristic peaks to 2-MI, indicating the retention of the imidazole ring after SL-Co-ZIF-L is formed. However, the broad peak at around  $1842\text{ cm}^{-1}$  corresponding to the stretching vibration of the N–H bond completely disappears, which demonstrates the formation of a coordination structure by the coordination interaction between  $\text{Co}^{2+}$  and N atoms in the –NH– group of 2-MI.<sup>44</sup> Besides, the FT-IR spectra of SL-Co-ZIF-L and ZIF-67 are remarkably semblable, verifying their similar chemical compositions. XPS measurement was further conducted to identify the chemical states of surface Co species. As revealed by Fig. S3,<sup>†</sup> surface Co exists in the form of  $\text{Co}^{2+}$  for both SL-Co-ZIF-L and ZIF-67. Moreover, the binding energy peak positions are also very close, indicating the similar chemical states of Co in these two samples.

In addition, the thermal stability of SL-Co-ZIF-L was characterized by TG analysis. As seen in TG-DTG curves (Fig. S4<sup>†</sup>), SL-Co-ZIF-L encounters two stages of rapid weight loss. The first appears between  $234\text{ }^{\circ}\text{C}$  and  $320\text{ }^{\circ}\text{C}$  with a DTG peak at  $292\text{ }^{\circ}\text{C}$ , which is mainly attributed to the removal of weakly linked 2-MI (the boiling point of 2-MI is around  $267\text{ }^{\circ}\text{C}$ ) in the structure of Co-ZIF-L,<sup>42,43</sup> while the second appears above  $450\text{ }^{\circ}\text{C}$ , which is corresponding to the structural collapse of Co-ZIF-L. However, it is evident that SL-Co-ZIF-L is relatively stable below  $234\text{ }^{\circ}\text{C}$ , only with a slow weight loss of less than 9%, which is mainly attributed to the loss of solvent molecules (*i.e.*,  $\text{H}_2\text{O}$ ) and organic species. This variation tendency in the TG curve is similar to that of Co-ZIF-L, which has been reported by Zhang's group.<sup>42</sup>

The BET surface area and pore structure of SL-Co-ZIF-L were obtained by nitrogen adsorption–desorption isotherm measurements. As shown in Fig. 3, both SL-Co-ZIF-L and ZIF-67 show a type I isotherm, typical of microporous MOF materials.<sup>45,46</sup> Nevertheless, the  $\text{N}_2$  absorption capacity of SL-Co-ZIF-L is much lower than that of ZIF-67, indicating its smaller surface area and pore volume. Indeed, the BET surface area and total pore volume of SL-Co-ZIF-L are, respectively,  $340.61\text{ m}^2\text{ g}^{-1}$  and  $0.1434\text{ cm}^3\text{ g}^{-1}$ , much lower than those of ZIF-67 ( $1661\text{ m}^2\text{ g}^{-1}$  and  $0.6703\text{ cm}^3\text{ g}^{-1}$ ). The huge difference in surface area between SL-Co-ZIF-L and ZIF-67 is in accordance with the observation from electron microscopy, where SL-Co-ZIF-L shows a much bigger size with a micro-sized and agglomerate morphology, while ZIF-67 is a submicro-sized rhombic dodecahedron with favourable dispersibility.

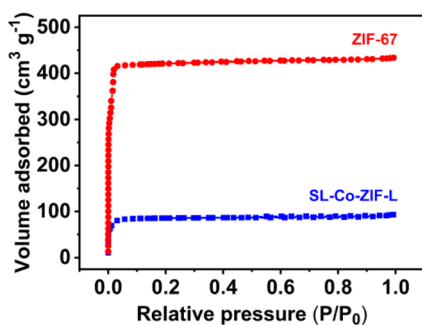


Fig. 3 Nitrogen adsorption–desorption isotherms of SL-Co-ZIF-L and ZIF-67.

## Catalytic performance towards the thermal decomposition of RDX

RDX is a typical high-energy additive broadly used in both solid propellants and gun propellants. In recent years, with the increase of RDX content in propellants for the sake of promoting energy level, the decomposition rate and efficiency of RDX are increasingly significant for sufficient energy release of RDX-containing propellants. Herein, the catalytic performance of SL-Co-ZIF-L towards the thermal decomposition of RDX was examined by differential scanning calorimetry (DSC). After thoroughly mixing the catalyst and RDX with a mass ratio of 1 : 4, the as-received compound was transferred to a pierced alumina crucible and heated with a heating rate of  $10\text{ }^{\circ}\text{C min}^{-1}$  under a nitrogen atmosphere. The obtained DSC curves (Fig. 4) can reflect the catalytic effect of combustion catalysts.

As seen in Fig. 4, pure RDX firstly undergoes a melting process, leading to an endothermic peak at  $205.2\text{ }^{\circ}\text{C}$ . Then a sluggish decomposition process appears successively, corresponding to the broad exothermic peak at  $242.1\text{ }^{\circ}\text{C}$ . After the import of SL-Co-ZIF-L, however, the DSC curve of RDX alters distinctly. The melting peak disappears, which means that the thermal decomposition of RDX takes place before melting with an initial temperature of about  $190.8\text{ }^{\circ}\text{C}$ ; thus, the exothermic decomposition peak will cover up the endothermic melting peak. This phenomenon is rarely seen for RDX decomposition catalysts. In addition, a sharp and intense decomposition peak appears much earlier at  $199.2\text{ }^{\circ}\text{C}$  and ends at around  $230.6\text{ }^{\circ}\text{C}$ . To the best of our knowledge, the decomposition peak temperature reduction ( $42.9\text{ }^{\circ}\text{C}$ ) after the addition of SL-Co-ZIF-L is obviously greater than that of all existing RDX/MOF compounds under similar test conditions (Table 1). Besides, the difference in peak shape also indicates that the decomposition of RDX might follow a different mechanism after mixing with SL-Co-ZIF-L. Moreover, the heat release calculated from the peak area increases significantly from  $1489\text{ J g}^{-1}$  to  $2245\text{ J g}^{-1}$ , indicating that the decomposition process is more thorough. To exclude the thermal decomposition effect of SL-Co-ZIF-L itself, the DSC curve of pure SL-Co-ZIF-L was also obtained under the same conditions, which demonstrates that the self-decomposition of SL-Co-ZIF-L has no contribution to the energy release of the RDX/SL-Co-ZIF-L compound as the decomposition process of RDX in the compound is already over

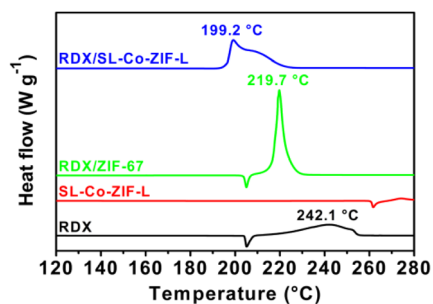


Fig. 4 DSC curves of RDX, SL-Co-ZIF-L, RDX/ZIF-67 and RDX/SL-Co-ZIF-L compounds.

**Table 1** The comparison of thermal decomposition peak temperature of RDX/MOF compounds using different MOFs

MOFs	$T_p^a$ (°C)	$\Delta T_p^b$ (°C)
SL-Co-ZIF-L	199.2	-42.9
ZIF-67	219.7	-22.4
[PbZn(TATT)(OH)(H <sub>2</sub> O)] <sub>n</sub> <sup>28</sup>	215.5	-24.8
[Co <sub>2</sub> (1-mbt) <sub>2</sub> (N <sub>3</sub> ) <sub>4</sub> ] <sub>n</sub> <sup>29</sup>	212	-34
[Pb(bta)(H <sub>2</sub> O)] <sub>n</sub> <sup>30</sup>	230.1	-10.1
[Ag <sub>2</sub> (5-ATZ)(N <sub>3</sub> )] <sub>n</sub> <sup>31</sup>	228.9	-17.5
[Ag <sub>3</sub> (H <sub>2</sub> ZTO) <sub>3</sub> (NO <sub>3</sub> ) <sub>2</sub> (CH <sub>3</sub> COO)] <sub>n</sub> <sup>47</sup>	213.5	-26.7
[Pb(Htztr) <sub>2</sub> (H <sub>2</sub> O)] <sub>n</sub> <sup>48</sup>	230	-16
[Pb <sub>2</sub> (C <sub>5</sub> H <sub>3</sub> N <sub>5</sub> O <sub>5</sub> ) <sub>2</sub> (NMP)·NMP] <sub>n</sub> <sup>49</sup>	246.2	-2.0

<sup>a</sup> Thermal decomposition peak temperature of RDX/MOF compounds.

<sup>b</sup> Change of thermal decomposition peak temperature of RDX/MOF compounds compared with pure RDX.

before the decomposition of SL-Co-ZIF-L (Fig. 4). As a comparison, the catalytic performance of ZIF-67 was examined as well. The decomposition peak of RDX/ZIF-67 appears at 219.7 °C, lower than that of RDX by 22.4 °C, but much higher than that of RDX/SL-Co-ZIF-L, although it seems more intense. It is worth mentioning that the two raw materials for the synthesis of SL-Co-ZIF-L, *i.e.*, Co(NO<sub>3</sub>)<sub>2</sub>·6H<sub>2</sub>O and 2-MI, almost have no facilitation to the thermal decomposition of RDX (Fig. S5†), revealing the unique role of the formed coordination polymer. By an overall consideration including the decomposition temperature and heat release, SL-Co-ZIF-L has an unprecedented catalytic effect towards the thermal decomposition of RDX, which is superior to almost all the reported catalysts and ZIF-67 under similar test conditions (Table S1†). Such an excellent catalytic effect will be beneficial to the energy release of RDX-containing propellants during combustion.

### Insight into the superior performance of SL-Co-ZIF-L

From the result of nitrogen adsorption-desorption isotherm measurements, SL-Co-ZIF-L has a much smaller BET surface area compared with ZIF-67. As a consequence, SL-Co-ZIF-L ought to have no advantages from the aspect of exposed catalytically active sites. This makes it quite interesting why SL-Co-ZIF-L embodies such a fascinating catalytic performance, especially considering that Co-ZIF-L and ZIF-67 are two types of Co-MOFs with the same ligand but different topological structures. So it is extremely important to gain deep insight into the superior performance of SL-Co-ZIF-L.

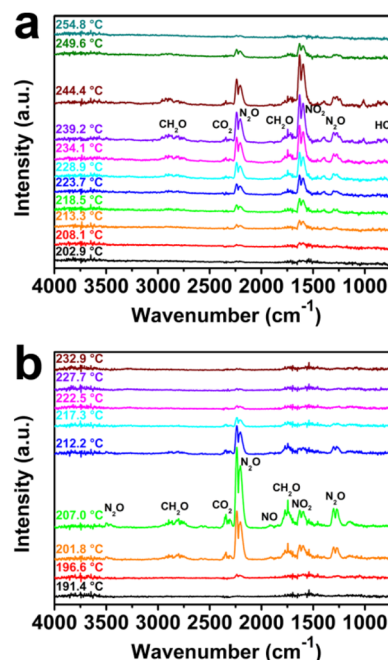
Non-isothermal kinetic analysis was firstly conducted for the purpose of obtaining the kinetic characteristics for the thermal decomposition process of RDX/SL-Co-ZIF-L. Both Kissinger's and Ozawa's methods were applied to estimate the corresponding kinetic parameters. The Kissinger equation and Ozawa equation are shown as eqn (1) and (2), respectively:

$$\ln \frac{\beta}{T_p^2} = \ln \frac{AR}{E} - \frac{E}{RT_p} \quad (1)$$

$$\log \beta + \frac{0.4567E}{RT_p} = C \quad (2)$$

where  $T_p$  is the thermodecomposition peak temperature,  $\beta$  is the linear heating rate,  $E$  is the apparent activation energy,  $A$  is the pre-exponential factor,  $R$  is the molar gas constant (8.314 J mol<sup>-1</sup> K<sup>-1</sup>), and  $C$  is a constant. Based on the thermodecomposition peak temperature of DSC measurements under different heating rates (5, 10, 15 and 20 °C min<sup>-1</sup>), the kinetic parameters including the apparent activation energy ( $E_k$ ,  $E_o$ ) and the pre-exponential factor ( $A_k$ ) were estimated by linear fitting of the two equations, respectively (subscript k stands for Kissinger's method, and o stands for Ozawa's method). The corresponding results are listed in Table S2,† where  $R_k$  and  $R_o$  are linear correlation coefficients in the fitted Kissinger equation and Ozawa equation, respectively. It can be found that the calculated  $E_k$  and  $E_o$  for both samples are close, and thus the apparent activation energy  $E_a$  can be calculated as the average of  $E_k$  and  $E_o$ . According to the above analysis,  $E_a$  for RDX/SL-Co-ZIF-L and pure RDX is 212.75 and 190.60 kJ mol<sup>-1</sup>, respectively. This phenomenon is correlated with the phase state of RDX when the decomposition is underway. While the thermal decomposition of pure RDX takes place after the fusion process and is largely dominated by the decomposition of its gas phase, the import of SL-Co-ZIF-L can facilitate the thermal decomposition of RDX early in its condensed phase (even in the crystal phase). It is evident that  $E_a$  for RDX decomposition in the condensed phase is always far higher than that in the gas phase, which is consistent with both theoretical and experimental studies.<sup>50,51</sup>

With this in mind, the decomposition mechanism of RDX/SL-Co-ZIF-L was deeply investigated by means of TG-DSC-FTIR-MS. As illustrated in Fig. 5, the *in situ* FT-IR spectra of gaseous products during the thermal decomposition of RDX/SL-

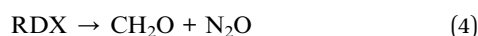


**Fig. 5** *In situ* FT-IR spectra of gaseous products during the thermal decomposition of (a) pure RDX and (b) RDX/SL-Co-ZIF-L.

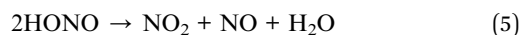


Co-ZIF-L and pure RDX show a marked difference. Corresponding with DSC curves, the thermal decomposition process of pure RDX starts early, perhaps below 208 °C, and lasts for a long period until more than 250 °C. The maximum decomposition velocity arises at around 240 °C. During the decomposition of RDX, the main gaseous products including NO<sub>2</sub> (1598 and 1630 cm<sup>-1</sup>), N<sub>2</sub>O (2204, 2239 and 1210–1330 cm<sup>-1</sup>), CH<sub>2</sub>O (2690–2980 and 1660–1820 cm<sup>-1</sup>), HCN (714 cm<sup>-1</sup>) and a small amount of CO<sub>2</sub> (2270–2390 cm<sup>-1</sup>) are slowly released. However, after the import of SL-Co-ZIF-L, the decomposition process of RDX changes a lot. First, the thermal decomposition of RDX/SL-Co-ZIF-L starts much earlier than RDX, perhaps below 196 °C, but lasts for a shorter period until more than 228 °C. Besides, the maximum decomposition velocity arises at around 207 °C, much earlier than that of RDX (around 240 °C). This means that when the decomposition of RDX/SL-Co-ZIF-L has completely finished, the decomposition rate of pure RDX is still far from the soonest, indicating the surprising catalytic performance of SL-Co-ZIF-L. Moreover, the IR peak of HCN at 714 cm<sup>-1</sup> almost disappears, while an inconspicuous IR peak of NO at 1909 cm<sup>-1</sup> appears, which demonstrates that the yields of typical gas products are also extremely different for the decomposition of RDX and RDX/SL-Co-ZIF-L. Furthermore, it is noticeable that the main peak height of NO<sub>2</sub> and N<sub>2</sub>O shows a hugely different characterization for RDX and RDX/SL-Co-ZIF-L. For RDX, the peak height of NO<sub>2</sub> at 1630 cm<sup>-1</sup> is always higher than that of N<sub>2</sub>O at 2239 cm<sup>-1</sup> during the whole decomposition process, with a N<sub>2</sub>O/NO<sub>2</sub> ratio of 0.65 at the maximum decomposition velocity. For RDX/SL-Co-ZIF-L, on the contrary, the peak height of N<sub>2</sub>O is far higher than that of NO<sub>2</sub>, which means that the ratio of N<sub>2</sub>O to NO<sub>2</sub> produced through the decomposition of RDX is much higher in this case (N<sub>2</sub>O/NO<sub>2</sub> ratio is 5.38 at the maximum decomposition velocity). Moreover, the more obvious peak of CH<sub>2</sub>O at 1660–1820 cm<sup>-1</sup> for RDX/SL-Co-ZIF-L indicates that CH<sub>2</sub>O will be produced more easily during the thermal decomposition of RDX/SL-Co-ZIF-L. In a word, the composition of gaseous products is quite different for RDX and RDX/SL-Co-ZIF-L, where there are more N<sub>2</sub>O and CH<sub>2</sub>O released for the decomposition of RDX/SL-Co-ZIF-L, while NO<sub>2</sub> is the most advantageous product for pure RDX.

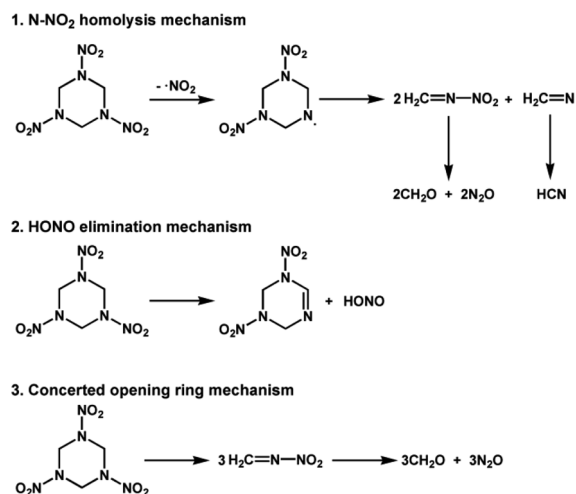
In addition, HCN (*m/z* = 27) is also observed as a product during the decomposition of pure RDX in the mass spectrum (Fig. S6†). Nevertheless, the peak is almost invisible for RDX/SL-Co-ZIF-L, which is consistent with the IR spectra. Combined with the results of FT-IR spectra, it can be concluded that after the import of SL-Co-ZIF-L, the main gaseous products during the decomposition of RDX will transform from NO<sub>2</sub> and HCN into N<sub>2</sub>O and CH<sub>2</sub>O. It is a consensus that the initial decomposition of RDX is a competition of N–N fission and C–N fission,<sup>13,51–53</sup> and the corresponding characteristic products are quite different, shown, respectively, as eqn (3) and (4):



To be specific, the mechanisms characterized by N–N fission mainly include the N–NO<sub>2</sub> homolysis mechanism and HONO elimination mechanism, where the characteristic products are HCN/HONO, while the mechanism characterized by C–N fission mainly refers to the concerted ring opening mechanism, where the characteristic products are N<sub>2</sub>O/CH<sub>2</sub>O (Scheme 1). During the N–N fission pathway, NO<sub>2</sub> can also be produced from quick decomposition of HONO, shown as eqn (5).<sup>13,52</sup>



As a consequence, N–N fission is dominant during the decomposition of pure RDX, judging from the main gaseous products, while C–N fission is dominant when SL-Co-ZIF-L is added. Another evidence is the enhanced heat release of RDX/SL-Co-ZIF-L (2245 J g<sup>-1</sup>) compared with pure RDX (1489 J g<sup>-1</sup>). As shown in eqn (3) and (4), N–N fission is an endothermic reaction, while C–N fission is exothermic. Thus there will be more heat released if C–N fission is dominant. It has to be pointed out that N–NO<sub>2</sub> homolysis and HONO elimination mechanisms often occupy the leading status under low-pressure and high-temperature conditions, especially in the gas-phase decomposition of RDX.<sup>51,54</sup> Considering that the decomposition of RDX usually begins after the fusion process and RDX is easy to vaporize, the N–N fission should be the dominant decomposition pathway in most cases.<sup>51,53–55</sup> Even for RDX in the crystal phase, there have been theoretical calculations proving that HONO elimination is the most favourable initial decomposition pathway owing to the lowest reaction barrier.<sup>51</sup> Conversely, C–N fission might only dominate during the decomposition of RDX in the condensed phase, which is relatively infrequent in relevant literature studies. Therefore, it is quite interesting that the decomposition of RDX/SL-Co-ZIF-L mainly follows the C–N fission pathway. Nevertheless, it is worth noting that the thermal decomposition of RDX takes place before melting when SL-Co-ZIF-L is added, which has



Scheme 1 Mainstream mechanisms for the thermal decomposition of RDX.



been inferred from the disappeared melting peak in the DSC curve (Fig. 4). This phenomenon exactly coincides with the mechanism analyses above. The decomposition of RDX in RDX/SL-Co-ZIF-L can take place early in the condensed phase (including the crystal phase), thus probably following the C–N fission mechanism.

From the analyses above, it is rational to speculate that the addition of SL-Co-ZIF-L can activate the condensed RDX and change the thermal decomposition mechanism from N–N fission to C–N fission. But there still remains the question why SL-Co-ZIF-L can exhibit far superior catalytic capability to ZIF-67. As shown in Fig. S7,† the *in situ* FT-IR spectra of gaseous products during the thermal decomposition of RDX/ZIF-67 reveal the feature which is intermediate between pure RDX and RDX/SL-Co-ZIF-L. In other words, the main gaseous products during the decomposition of RDX/ZIF-67 contain more N<sub>2</sub>O and CH<sub>2</sub>O rather than NO<sub>2</sub> and HCN (peak height ratio of N<sub>2</sub>O/NO<sub>2</sub> is 0.90) compared with pure RDX, indicating that the decomposition pathway of RDX begins to transform from N–N fission to C–N fission after the introduction of ZIF-67. This phenomenon is also in accordance with the intermediate decomposition peak reflected in the DSC curves (Fig. 4). Nevertheless, on the one hand, SL-Co-ZIF-L and ZIF-67 almost have the same chemical compositions and surface Co species, as reflected by the aforementioned characterization. On the other hand, ZIF-67 with a submicro-sized and well-dispersed morphology shows a much larger BET surface area and total pore volume than the micro-sized SL-Co-ZIF-L. So it is spontaneous to treat ZIF-67 as a better catalyst on account of the more exposed catalytically active sites. However, the current fact is surprisingly opposite.

Knocking out the factors of chemical composition, surface Co species and specific surface area, it is now necessary to give a deep insight into the different crystalline structures of ZIF-67 and Co-ZIF-L. Although ZIF-67 with a 3D porous structure has a much larger surface area, its rigid construction and narrow micropores (Fig. S8†) make it difficult to accommodate the large RDX molecule with a six-membered heterocyclic ring and extended N–NO<sub>2</sub> groups. This hinders the contact between RDX and the internal structure of ZIF-67, thus making these micropores “useless” and leading to the sharp decrease of accessible active sites during the thermal decomposition processes. On the contrary, Co-ZIF-L has a 2D layered structure, which is different from ZIF-67 with a 3D sodalite topology, just like the relationship between Zn-ZIF-L and ZIF-8.<sup>42,43</sup> Specifically, the 2D layers of Co-ZIF-L are part of the sodalite topology found in the 3D structure of ZIF-67 (Fig. S9†). The adjacent 2D layers are mainly stabilized by two types of interactions. One is the hydrogen bonds between terminal 2-MI ligands and the “free” 2-MI between the layers, and the other is the interactions between “free” 2-MI and another parallel 2-MI in the neighbouring layer (Fig. S10†). As a consequence, it is predictable that such weak interactions will be easily broken upon heating during the testing process of catalytic thermal decomposition of RDX. Thus the 2D layers of Co-ZIF-L will be separated and exposed to RDX molecules and reaction intermediates, which means that the originally “hidden” catalytic sites will be largely accessible

to some extent. In a word, although SL-Co-ZIF-L has a smaller specific area and also narrow micropores under general conditions, the true catalytically active sites of SL-Co-ZIF-L in the dynamic testing environment are more than ZIF-67 instead.

From another perspective, SL-Co-ZIF-L was prepared in pure H<sub>2</sub>O at room temperature. Therefore, the coordination sites of Co<sup>2+</sup> in SL-Co-ZIF-L were likely to be occupied by some weakly coordinated H<sub>2</sub>O molecules.<sup>56,57</sup> This can be inferred from TG-DTG analyses to a certain extent. In the TG-DTG curves of ZIF-67 (Fig. S11†), it is clear that ZIF-67 shows a slow weight loss between 92 °C and 185 °C, which is mainly ascribed to the loss of residual H<sub>2</sub>O and organic species. In contrast, SL-Co-ZIF-L persistently loses weight in the higher temperature range until the first sharp loss starting at 234 °C (Fig. S4†), with a DTG peak at 181 °C, which is also higher than that of ZIF-67 (162 °C). In addition, the weight loss of SL-Co-ZIF-L at this stage is also more than that of ZIF-67 although its surface area is much smaller. These mean that SL-Co-ZIF-L might contain some coordinated H<sub>2</sub>O molecules in addition to the easily removed adsorbed H<sub>2</sub>O. So it can be predicted that the coordinated H<sub>2</sub>O molecules will be removed with the elevated temperature during the catalytic decomposition process of RDX, thus leading to the exposure of some unsaturated coordination sites. To illustrate the impact of coordination sites of Co-ZIFs towards their catalytic activity and mechanism, DFT calculations were further conducted. For the sake of simplifying computational models, the normal four-coordinated Co<sup>2+</sup> was used to represent ZIF-67, while Co<sup>2+</sup> coordinated by two 2-MI molecules was

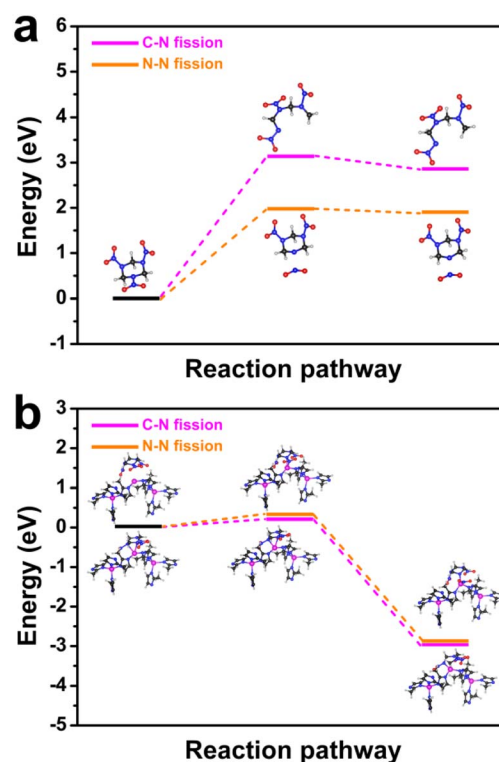


Fig. 6 Free energy for N–N fission and C–N fission decomposition pathways of (a) pure RDX and (b) RDX/SL-Co-ZIF-L.



selected to represent Co-ZIF-L with unsaturated coordination sites. For the initial decomposition process of pure RDX (shown in Fig. 6a), the reaction barrier for the N–N fission pathway is calculated to be 1.97 eV and the corresponding free energy of the final state is 1.91 eV; both are much lower than those for the C–N fission pathway (3.15 eV and 2.86 eV, respectively). This is consistent with existing literature studies and the general understanding that N–N fission is more advantageous in most cases.<sup>51,53–55</sup> When ZIF-67 is used as the catalyst, the four-coordinated Co<sup>2+</sup> will become a catalytic site that can activate both C–N fission and N–N fission processes. To be specific, after the adsorption and activation effect of ZIF-67, the activation energy of the N–N fission pathway will be lowered by 1.39 eV (from 1.97 eV to 0.58 eV), while the activation energy of the C–N fission pathway will also be reduced from 3.15 eV to 1.84 eV (Fig. S12†). What is more, the final state for both N–N fission and C–N fission pathways can be stabilized by ZIF-67. Nevertheless, the reaction priority is not changed, which means that the N–N fission pathway is still the more advantageous pathway. However, when Co-ZIF-L is used as the catalyst, the unsaturated coordination sites will become a totally different active site. After the activation effect of Co-ZIF-L, the C–N bond is stretched from 1.47 Å to 1.66 Å, which means that the C–N bond is sufficiently activated and easy to break. Moreover, the energy changes during the N–N or C–N fission pathway are distinctly different from the above two situations. As revealed from Fig. 6b, although the reaction barrier of N–N fission pathway is reduced from 1.97 eV to 0.31 eV when using Co-ZIF-L as the catalyst, the reaction barrier of the C–N fission pathway is largely reduced from 3.15 eV to 0.22 eV, indicating that C–N fission seems to be more dominant than N–N fission. In addition, the final state energy of the C–N fission pathway (–2.93 eV) is also lower than that of the N–N fission pathway (–2.88 eV). This reversal makes C–N fission a more dominant pathway from the perspective of free energy, which is in accordance with the results of catalytic decomposition experiments above and confirms the unique role of SL-Co-ZIF-L as a superior catalyst for the thermal decomposition of RDX by facilitating the generally disadvantageous C–N fission pathway.

## Conclusions

In conclusion, star-like Co-ZIF-L (SL-Co-ZIF-L) with a micron size was synthesized through a room-temperature and aqueous-phase reaction, which exhibited unprecedented catalytic capability for the thermal decomposition of RDX. Concretely, SL-Co-ZIF-L can lower the decomposition temperature of RDX by 42.9 °C and boost the heat release by 50.8%, which is superior to most ever-reported catalysts including energetic MOFs. Interestingly, SL-Co-ZIF-L also showed far better performances than ZIF-67, which has almost the same chemical compositions but a much larger surface area. In-depth mechanism study from both experimental and theoretical perspectives reflects the reasons that the import of SL-Co-ZIF-L with a weakly interacted 2D structure could not only expose more catalytically active sites by the heating effect during the catalytic process, but also activate the exothermic C–N fission pathway for the

decomposition of RDX in the condensed phase, thus reversing the commonly advantageous N–N fission pathway and promoting the decomposition process in the low-temperature stage. Our research reflects and dissects the abnormal catalytic performance of micro-sized MOF catalysts with a 2D layered structure, which will provide guidance for the rational structure design of catalysts used in micromolecule transformation reactions, *e.g.*, the thermal decomposition of energetic materials.

## Conflicts of interest

There are no conflicts to declare.

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## References

- 1 T. An, F. Q. Zhao, Q. L. Yan, Y. J. Yang, Y. J. Luo, J. H. Yi and W. L. Hong, *Propellants, Explos., Pyrotech.*, 2018, **43**, 1087–1095.
- 2 F. Zhen, X. Y. Zhou, L. Q. Wang, R. J. Yang and F. L. Huang, *Propellants, Explos., Pyrotech.*, 2019, **44**, 362–368.
- 3 L. Liang, X. Guo, Z. Liang and G. Li, *J. Energ. Mater.*, 2018, **36**, 339–351.
- 4 D. R. Kshirsagar, S. Jain, S. N. Jawalkar, N. H. Naik, S. Pawar and M. Maurya, *Propellants, Explos., Pyrotech.*, 2016, **41**, 304–311.
- 5 W. Wang, T. Wei, H. Gao, L. Xiao, K. Xu and F. Zhao, *Chin. J. Explos. Propellants*, 2017, **40**, 29–35.
- 6 T. Wei, Y. Zhang, K. Xu, Z. Ren, H. Gao and F. Zhao, *RSC Adv.*, 2015, **5**, 70323–70328.
- 7 W. Q. Pang, L. T. DeLuca, X. Z. Fan, O. G. Glotov, K. Wang, Z. Qin and F. Q. Zhao, *Combust. Flame*, 2020, **220**, 157–167.
- 8 F. Zhen, X. Y. Zhou, L. Q. Wang, R. J. Yang and F. L. Huang, *Propellants, Explos., Pyrotech.*, 2019, **44**, 362–368.
- 9 H. Li, B. Liu, Y. Xu, C. Wang, Z. Sun, H. Li, E. Yao, J. Yi, Z. Qin and F. Zhao, *J. Anal. Appl. Pyrolysis*, 2021, **157**, 105228.
- 10 D. W. Ren, M. S. Wei, J. H. Shi, B. D. Wu, Y. Li and J. Y. Wang, *Z. Anorg. Allg. Chem.*, 2021, **648**, e202100106.
- 11 J. Shen, Z. Liu, B. Xu, F. Chen, Y. Zhu, Y. Fu, D. J. Kline, X. Liao and Z. Wang, *J. Energ. Mater.*, 2020, **38**, 326–335.
- 12 J. Shen, Z. Liu, B. Xu, H. Liang, Y. Zhu, X. Liao and Z. Wang, *Propellants, Explos., Pyrotech.*, 2019, **44**, 355–361.
- 13 R. Xu, L. Yang, H. Fang, Z. Mi, Y. Lv, L. Jiang, X. Fu, J. Li and G. Zhang, *ACS Appl. Nano Mater.*, 2022, **5**, 14942–14953.
- 14 J. Wang, X. Lian, S. Chen, H. Li and K. Xu, *J. Colloid Interface Sci.*, 2022, **610**, 842–853.
- 15 J. Wang, X. Lian, Q. Yan, D. Gao, F. Zhao and K. Xu, *ACS Appl. Mater. Interfaces*, 2020, **12**, 28496–28509.
- 16 S. Hanafi, D. Trache, W. He, W. X. Xie, A. Mezroua and Q. L. Yan, *J. Phys. Chem. C*, 2020, **124**, 5182–5195.
- 17 H. Mei, Y. Xu, G. Lei, W. Cao, Z. Li and J. Zhang, *New J. Chem.*, 2022, **46**, 1687–1692.



- 18 C. Y. Wu, Y. C. Chen, T. H. Wang, C. I. Lin, C. W. Huang and S. R. Zhou, *Combust. Flame*, 2021, **232**, 111556.
- 19 J. Zhang, B. Jin, X. Li, W. Hao, T. Huang, B. Lei, Z. Guo, J. Shen and R. Peng, *Chem. Eng. J.*, 2021, **404**, 126287.
- 20 Y. Liu, S. Jin, H. Yang, S. Li, W. Xie, Y. Zhao, W. Zhang, Y. Chen and X. Fan, *Combust. Flame*, 2021, **225**, 57–64.
- 21 M. Rimoldi, A. J. Howarth, M. R. DeStefano, L. Lin, S. Goswami, P. Li, J. T. Hupp and O. K. Farha, *ACS Catal.*, 2017, **7**, 997–1014.
- 22 A. J. Howarth, Y. Liu, P. Li, Z. Li, T. C. Wang, J. T. Hupp and O. K. Farha, *Nat. Rev. Mater.*, 2016, **1**, 15018.
- 23 Y. Yang, Y. Bai, F. Zhao, E. Yao, J. Yi, C. Xuan and S. Chen, *RSC Adv.*, 2016, **6**, 67308–67314.
- 24 J. Yan, H. Wang, B. Jin, M. Zeng and R. Peng, *J. Solid State Chem.*, 2021, **297**, 122060.
- 25 Y. Huang, R. Peng and B. Jin, *New J. Chem.*, 2021, **45**, 11068–11074.
- 26 S. Wu, M. Li, Z. Yang, Z. Xia, B. Liu, Q. Yang, Q. Wei, G. Xie, S. Chen, S. Gao and J. Y. Lu, *Green Chem.*, 2020, **22**, 5050–5058.
- 27 Q. Yang, G. Yang, W. Zhang, S. Zhang, Z. Yang, G. Xie, Q. Wei, S. Chen and S. Gao, *Chem. - Eur. J.*, 2017, **23**, 9149–9155.
- 28 S. Wu, G. Lin, Z. Yang, Q. Yang, Q. Wei, G. Xie, S. Chen, S. Gao and J. Y. Lu, *New J. Chem.*, 2019, **43**, 14336–14342.
- 29 X. Ma, Y. Liu, W. Song, Z. Wang, X. Liu, G. Xie, S. Chen and S. Gao, *Dalton Trans.*, 2018, **47**, 12092–12104.
- 30 Q. Yang, X. Song, W. Zhang, L. Hou, Q. Gong, G. Xie, Q. Wei, S. Chen and S. Gao, *Dalton Trans.*, 2017, **46**, 2626–2634.
- 31 X. Qu, Q. Yang, J. Han, Q. Wei, G. Xie, S. Chen and S. Gao, *RSC Adv.*, 2016, **6**, 46212–46217.
- 32 Y. Liu, P. Yi, L. Gong, X. Yi, P. He, T. Wang and J. Zhang, *Inorg. Chem.*, 2023, **62**, 3186–3194.
- 33 C. Liang, J. Yang, Y. Guo, F. Yuan, W. Song, J. Xi, X. Liu and S. Chen, *Inorg. Chem.*, 2021, **60**, 9282–9286.
- 34 S. Cao, X. Ma, X. Ma, P. Cen, Y. Wu, J. Yang, X. Liu, G. Xie and S. Chen, *Dalton Trans.*, 2020, **49**, 2300–2307.
- 35 J. Zhao, X. Li, M. Zhang, Z. Xu, X. Qin, Y. Liu, L. Han and G. Li, *Nanoscale*, 2023, **15**, 4612–4619.
- 36 H. Zhang, S. Wang, M. Wang, G. Li, L. Yu, X. Liu, Z. Wang and C. Zhang, *J. Rare Earths*, 2022, DOI: [10.1016/j.jre.2022.12.006](https://doi.org/10.1016/j.jre.2022.12.006).
- 37 J. Wang, Y. Li, S. Xu, C. Lin, X. Ma, Y. Ni and S. Cao, *New J. Chem.*, 2022, **46**, 2962–2970.
- 38 K. Wang, S. Liu, Y. Li, G. Wang, M. Yang and Z. Jin, *Appl. Surf. Sci.*, 2022, **601**, 154174.
- 39 J. Mor, S. K. Sharma, P. Utpalla, J. Bahadur, J. Prakash, A. Kumar and P. K. Pujari, *Microporous Mesoporous Mater.*, 2022, **335**, 111814.
- 40 Y. He, Z. Abedi, C. Ni, S. Milliken, K. M. O'Connor, D. G. Ivey and J. G. C. Veinot, *ACS Appl. Nano Mater.*, 2022, **5**, 12496–12505.
- 41 B. Zhang, H. Xu, Q. Chen, H. Chen and G. He, *J. Alloys Compd.*, 2022, **922**, 166216.
- 42 J. Zhang, T. Zhang, D. Yu, K. Xiao and Y. Hong, *CrystEngComm*, 2015, **17**, 8212–8215.
- 43 R. Chen, J. Yao, Q. Gu, S. Smeets, C. Baerlocher, H. Gu, D. Zhu, W. Morris, O. M. Yaghi and H. Wang, *Chem. Commun.*, 2013, **49**, 9500–9502.
- 44 M. Wang, J. Liu, C. Guo, X. Gao, C. Gong, Y. Wang, B. Liu, X. Li, G. G. Gurzadyana and L. Sun, *J. Mater. Chem. A*, 2018, **6**, 4768–4775.
- 45 N. T. Tran, L. G. Trung and M. K. Nguyen, *J. Solid State Chem.*, 2021, **300**, 122287.
- 46 M. Jafarinasab, A. Akbari, M. Omidkhah and M. Shakeri, *Energy Fuels*, 2020, **34**, 12779–12791.
- 47 X. Song, S. Zhang, G. Zhao, W. Zhang, D. Chen, Q. Yang, Q. Wei, G. Xie, D. Yang, S. Chen and S. Gao, *RSC Adv.*, 2016, **6**, 93231–93237.
- 48 W. Gao, X. Liu, Z. Su, S. Zhang, Q. Yang, Q. Wei, S. Chen, G. Xie, X. Yang and S. Gao, *J. Mater. Chem. A*, 2014, **2**, 11958–11965.
- 49 J. J. Liu, Z. L. Liu, J. Cheng and D. Fang, *J. Solid State Chem.*, 2013, **200**, 43–48.
- 50 G. T. Long, S. Vyazovkin, B. A. Brems and C. A. Wight, *J. Phys. Chem. B*, 2000, **104**, 2570–2574.
- 51 Y. Ma, M. Lv, F. Shang, C. Zhang, J. Liu and P. Zhou, *J. Phys. Chem. A*, 2022, **126**, 1666–1673.
- 52 C. Wan, J. Li, S. Chen, W. Wang and K. Xu, *J. Anal. Appl. Pyrolysis*, 2021, **160**, 105372.
- 53 X. Chen and C. F. Goldsmith, *Proc. Combust. Inst.*, 2019, **37**, 3167–3173.
- 54 R. W. Molt Jr., T. Watson Jr., A. P. Bazanté, R. J. Bartlett and N. G. J. Richards, *Phys. Chem. Chem. Phys.*, 2016, **18**, 26069–26077.
- 55 M. J. Swadley and T. Li, *J. Chem. Theory Comput.*, 2007, **3**, 505–513.
- 56 Z. Zou, T. Wang, X. Zhao, W. J. Jiang, H. Pan, D. Gao and C. Xu, *ACS Catal.*, 2019, **9**, 7356–7364.
- 57 H. G. Hao, Y. F. Zhao, D. M. Chen, J. M. Yu, K. Tan, S. Ma, Y. Chabal, Z. M. Zhang, J. M. Dou, Z. H. Xiao, G. Day, H. C. Zhou and T. B. Lu, *Angew. Chem., Int. Ed.*, 2018, **57**, 16067–16071.

