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# Enhanced electrochromic properties of WO<sub>3</sub>/ITO nanocomposite smart windows

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Tungsten oxide is regarded as the most promising electrochromic material owing to its continuously tunable optical properties, low cost, and high coloration efficiency. Further improving its optical modulation, switching speed, and coloration efficiency is important to electrochromic smart windows and related devices. Here, we demonstrate an enhanced electrochromic film composed of a WO<sub>3</sub> nanosheet and ITO nanoparticles developed by an all-solution technology. The WO<sub>3</sub> nanosheet is fabricated by an acid-assisted hydrothermal process with high product efficiency. The introduction of an ITO into the WO<sub>3</sub> nanosheets significantly improved the electrochemical activity and the conductivity of the composite film. Compared with a reported electrochromic film without ITO doping, our synthesized composite WO<sub>3</sub> film exhibited optical modulation up to 88% and a high coloration efficiency of 154.16 cm<sup>2</sup> C<sup>-1</sup>. Particularly, our electrochromic film was based on the dispersant solution and spin-coating technology, which may also be realized with nano-spray coating for large scale applications. The results offer an effective way to develop large-area electrochromic film and devices.

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## 1. Introduction

Electrochromic (EC) devices can change in reflectance and transmittance in the range of visible and near infrared wavelength regions based on the redox reaction as a response to an external voltage. It has received much attention in the smart windows, sensors, and energy storage technologies,<sup>1–3</sup> *etc.* Tungsten oxide (WO<sub>3</sub>) is the most promising electrochromic material due to its outstanding optical properties, including fast switching speed, high coloration efficiency, and low-cost preparation process.<sup>4–6</sup> Currently, the major problem in the field of electrochromic materials<sup>7–9</sup> and devices is to further improve the contrast, coloration efficiency (CE), switching speed, and cycling life stability. In the past decades, researchers have proposed a number of strategies, such as constructing nanostructure,<sup>10,11</sup> regulating crystallinity,<sup>12,13</sup> and doping modifications,<sup>14,15</sup> to improve the EC performance based on WO<sub>3</sub>. For example, various nanostructured WO<sub>3</sub>, such as nanodots,<sup>16</sup> nanowires,<sup>17</sup> nanorods,<sup>18</sup> and nanosheets,<sup>19,20</sup> have been fabricated by liquid phase method. Yao *et al.* prepared a WO<sub>3</sub> quantum-dot (QD) film by a common atmospheric pressure solution-based deposition (APSD)

method, showing a huge optical modulation in the visible and near-infrared range.<sup>21</sup> Lavi *et al.* proposed an effective method to enhance tungstate oxide EC *via* the formation of a transparent conductive network by adding indium tin oxide (ITO) nanoparticles onto a WO<sub>3</sub> film.<sup>22</sup> Recently, Gao *et al.* experimentally showed higher optical modulation and faster switching speed by preparing hexagonal WO<sub>3</sub> nanorod and ITO composite films.<sup>23</sup> The architecture of ITO nanocrystals embedded in amorphous WO<sub>3</sub> with improved electrochromic performance was reported by Zhang *et al.*<sup>24</sup>

Compared with the one-dimensional (1D) WO<sub>3</sub> nanostructure, two-dimensional (2D) nanostructures, such as nanosheets and nanoflakes, have a larger surface area and stability, indicating more active electrochemical property and better EC characteristics. Most recently, a strategy of compositing amorphous WO<sub>3</sub> on crystalline WO<sub>3</sub> nanosheets<sup>14</sup> was proposed to improve the cyclic stability of EC devices. A. Azam *et al.* reported a novel solution-phase synthesis of 2D WO<sub>3</sub> nanosheets<sup>19</sup> through oxidation from 2D tungsten disulfide (WS<sub>2</sub>) nanosheets, exhibiting a greater color modulation of 62.57% at 700 nm wavelength compared to the conventional device and an enhancement of the response time. However, the properties, including optical modulation, coloration efficiency, and response time, for such EC devices based on WO<sub>3</sub> nanosheets could be further improved.

Here, we prepared monoclinic WO<sub>3</sub> nano-sheet by hydrothermal method and coated WO<sub>3</sub>/ITO composite film through spin coating on the ITO glass substrate. Remarkably, larger optical modulation, faster switching speed, and higher coloration efficiency are observed by adding ITO particles into the as-

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prepared monoclinic  $\text{WO}_3$  nano-sheet film. Our experimental results provide another way to develop EC devices and smart windows with excellent optical properties.

## 2. Experimental section

### 2.1 Preparation of $\text{WO}_3$ nanosheets

$\text{WO}_3$  nanosheets were synthesized with assisted ( $\text{H}_2\text{SO}_4$ ) hydrothermal process under a relatively low temperature acid. The main chemicals used for the present work such as  $\text{Na}_2\text{WO}_4 \cdot 2\text{H}_2\text{O}$ ,  $\text{AlCl}_3 \cdot 6\text{H}_2\text{O}$  are AR grades from Aladdin Reagent (Shanghai) Co., Ltd. Ethanol and PEG-600 were brought from Chinese Medicine Group Chemical Reagent Co., Ltd. The dispersant used in this work, BYK-2013, was purchased from BYK-Chemie® Inc. ITO nanoparticles and glasses with ITO were purchased from Nangong Bole Metal Materials Co., Ltd., and Luoyang Shangzhuo Technology Co., Ltd., respectively. All chemical reagents used in the experiment were of analytical grade without further purification.

In a typical synthesis of  $\text{WO}_3$  nanosheets,<sup>25</sup> 2.64 g  $\text{Na}_2\text{WO}_4 \cdot 2\text{H}_2\text{O}$  was firstly dissolved in 80 ml distilled water contained in a 100 ml beaker with 10 minutes stirring under room temperature, then 80 ml  $\text{H}_2\text{SO}_4$  (98%) was dropped into the above solution under vigorous stirring. In this process, the solution gradually changed color from transparent to yellow, indicating that  $\text{H}_2\text{WO}_4$  was formed. Then, the solution was heated to 160 °C for 4.5 h in an oil bath. After cooling to room temperature, the prepared product powders were centrifuged and washed alternately with distilled water and ethanol 3 times, respectively. Finally, the  $\text{WO}_3$  powder was dried at 60 °C for 12 h and the yellow  $\text{WO}_3$  nanosheet powder was obtained.

### 2.2 Preparation of the $\text{WO}_3$ /ITO dispersant solution and EC composite film

Firstly, 0.27 g polyethylene glycol and BYK2013 (0.18 g) were added into an ethanol solution (8.55 g), then the  $\text{WO}_3$  nanopowders and ITO nanoparticles were weighed according to the composition of Y (Y = ITO/ $\text{WO}_3$ , 0%, 8%, 16% and 24% in weight), respectively. The total amount of ITO and  $\text{WO}_3$  is fixed at around 10 g. Then, the mixture was milled with 0.5 mm  $\text{ZrO}_2$  beads for 6 h at 1000 rpm, producing the dispersed mixture of  $\text{WO}_3$  and ITO nanoparticles.

The glass substrates (25 mm × 12.5 mm × 1.6 mm) with ITO were cleaned under ultrasonic process by isopropyl alcohol, acetone, and deionized water, respectively. After the cleaning process, the ITO glasses were dried for several minutes.  $\text{WO}_3$  nanosheet dispersions (200  $\mu\text{l}$  each sample) with different ITO content were coated on processed ITO glass (ITO film/glass) by spin-coating method with a 4000 rpm rate. At last, all of the samples were annealed at 300 °C for 1 h to form better contact between composite films and ITO substrates. The prepared transparent EC films with the 0%, 8%, 16% and 24% ITO are denoted according to the following nomenclature, Y0, Y8, Y16 and Y24, respectively.

The schematic of the measurement experimental setup is shown in Fig. 1(a). In the process of measurement, the quartz

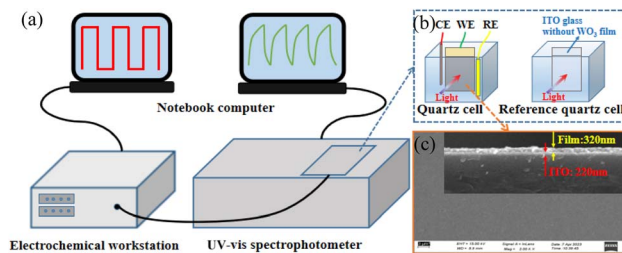


Fig. 1 (a) Schematic representation of the experimental setup, (b) quartz cell, and (c) the top-view and cross-sectional (inset) SEM images of the  $\text{WO}_3$ /ITO thin film.

cell and reference quartz cell shown in Fig. 1(b), were placed into a UV-Vis spectrophotometer.<sup>7,10,26</sup> At the same time, the square wave power was supplied by the electrochemical workstation from  $-0.7$  V to  $+0.7$  V. The transmittance of the thin film shown in Fig. 1(c) was recorded by a UV-Vis spectrophotometer and processed by a computer.

### 2.3 Characterization

The crystal phase of the samples was measured by X-ray diffraction (XRD) (3 kW, D8 ADVANCE, Bruker Corporation, Germany) using a copper K-alpha source. The surface morphology of the  $\text{WO}_3$ /ITO films after annealing were characterized by scanning electron microscopy (SEM, Sigma 300, Zeiss Oberkochen, Germany). The optical transmittance of films was measured by UV-Vis spectrophotometer (UV1901PC, Shanghai AuCy Scientific Instrument, Shanghai). The *in situ* spectroelectrochemical measurements were carried out in a quartz cell containing 1 M  $\text{AlCl}_3$  in  $\text{H}_2\text{O}$ <sup>20,27,28</sup> placed in an electrochemical workstation (CS 350, Wuhan Corrtest Instrument, Wuhan) using a three-electrode system consisting of a standard Ag/AgCl electrode as the reference electrode, a platinum plate as the counter electrode, and the fabricated  $\text{WO}_3$  films on ITO glass as the working electrode. Electrochemical impedance spectroscopy (EIS) measurements were conducted on an electrochemical workstation by applying an amplitude sinusoidal voltage of 10 mV in the frequency range of 0.01–100 kHz. The transmittance, optical modulation, and cycling stability measurement were carried out by combining a UV-Vis spectrophotometer and electrochemical workstation.

## 3. Results and discussion

Fig. 2(a) and (b) show the measured XRD patterns of the prepared  $\text{WO}_3$  nanosheets with 16% ITO content and its field emission scanning electron microscopy (SEM) image. The two lines displayed at the bottom of Fig. 2(a) correspond to  $\text{WO}_3$  (JCPDS no. 87-2385) and  $\text{In}_2\text{O}_3$  (JCPDS no. 65-3170), respectively. The blue and red curves denote the XRD spectrophotometer for the  $\text{WO}_3$  nanosheet powder with ITO contents of 0% and 16%, respectively. The Bragg peaks of the red and blue curves clearly show the monoclinic  $\text{WO}_3$  crystal faces indexed as (0 0 2), (1 1 0), (0 1 2), (2 0 0), (1 1 2), (0 2 2), (2 2 0), (3 1 0). Comparing



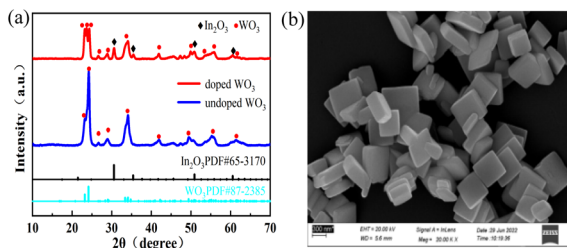


Fig. 2 (a) The measured XRD pattern of the as-prepared WO<sub>3</sub> nanosheet powder and WO<sub>3</sub>/ITO EC film. (b) SEM image of WO<sub>3</sub> with doped 16% ITO nanoparticles.

the curves shown in Fig. 2(a), it is clear that ITO has been successfully mixed into the monoclinic WO<sub>3</sub> powder. Fig. 2(b) exhibits the square morphology of the ITO-doped WO<sub>3</sub> nanosheets with size distributions of about 200–500 nm and thickness of around 100 nm.

To better understand the electrochemical behavior, we performed cyclic voltammetry (CV) and impedance (EI) spectroscopy measurements, shown in Fig. 3(a) and (b), respectively. The CV measurement was performed in 1 M AlCl<sub>3</sub>·H<sub>2</sub>O at a sweep rate of 100 mV s<sup>−1</sup> from −0.7 V to +0.7 V. As illustrated in Fig. 3(a), the cathodic current peaks are observed and the transparent films gained coloration with the reduction of W<sup>6+</sup> → W<sup>5+</sup> due to the Al<sup>3+</sup> ion insertion in the films. Reversing the potential from −0.7 V to +0.7 V, these processes correspond to the oxidation of W<sup>5+</sup> → W<sup>6+</sup>. In Fig. 2(a), the WO<sub>3</sub>/ITO nanocomposite films demonstrate a much larger area of CV and higher current density than the pure WO<sub>3</sub> films (Y0). The areas enclosed by the CV curves become larger with increasing ITO concentration, indicating that more ions and electrons are involved at the interface between the WO<sub>3</sub>/ITO nanocomposite films and the electrolyte. The CV measurement illustrates an enhancement of the electrochemical activity related to the EC performances, originating from the introduction of ITO into the nanosheet structures with higher surface area than that of the WO<sub>3</sub> nanorod.

Further evidence for the increased electric activity is demonstrated in Fig. 3(b) through the electrochemical impedance spectroscopy (EIS) measurements (Nyquist plot) by applying an amplitude sinusoidal voltage of 10 mV in the frequency range of 0.01–100 kHz. The curves show a semi-circle in high (inset of Fig. 3(b)) and skew cure in the low-frequency

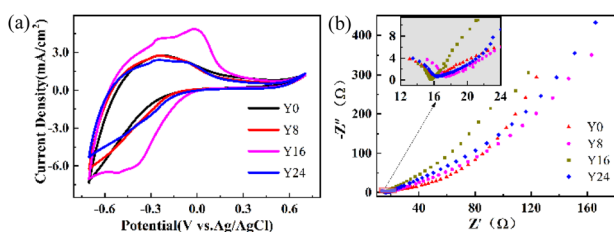
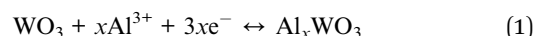


Fig. 3 (a) CV curves of the samples measured in the potential range of −0.7 V to 0.7 V at a scan rate of 100 mV s<sup>−1</sup> with a three-electrode system. (b) Nyquist plot of samples Y0, Y8, Y16, and Y24.

ranges. The slope of the curve in Fig. 3(b) corresponds to the ion diffusion in the process of intercalation/deintercalation of Al<sup>3+</sup>. The increase of the slope for samples Y0, Y8, Y16, and Y24 suggests that the conductivity is further increased<sup>23–25</sup> due to the increased ITO concentration from Y0 to Y24, agreeing with the measured CV curves in Fig. 3(a).

Fig. 4(a) and (b) illustrate the optical modulation for the prepared WO<sub>3</sub> nanosheet with ITO concentrations varying from 0% to 24% (denoted with Y0, Y8, Y16, Y24) at the bleached and colored states in the optical wavelength range of 400–1100 nm. The glasses with ITO substrate have become more transparent (high transmission) in the bleached state with the addition of ITO nanoparticles into the WO<sub>3</sub> nanosheet film. On the contrary, the composite WO<sub>3</sub>/ITO showed lower transmission in the colored state, except for sample Y24. Particularly, as shown in Fig. 4(b), the optical modulation rate  $\Delta T$  is up to 87.89% at wavelength 633 nm, which is higher than many reported electrochromic devices.<sup>10,29–31</sup> As illustrated in Table 1, the optical modulation has reached 90.28% around wavelength 749 nm. Fig. 4(c) plots the variation of the optical modulation  $\Delta T$  with the content of ITO, suggesting that the optimized concentration of ITO in the WO<sub>3</sub> nanosheet composite film is 16%. This optimized content of ITO is smaller than that of the reported 20% in the WO<sub>3</sub> nanorod and ITO composite film<sup>23</sup> due to the larger surface area in the prepared nanosheet morphology. The bleached (transparent) and colored (blue) states of the composite film with 16% ITO content are illustrated in Fig. 4(d), exhibiting the high contrast in the visible range. The switching process between the colored state and bleached state of the WO<sub>3</sub>/ITO films can be explained by the following EC reaction:<sup>27,28</sup>



We now turn to the coloration and bleaching time (switching speed), which is an important criterion for the practical

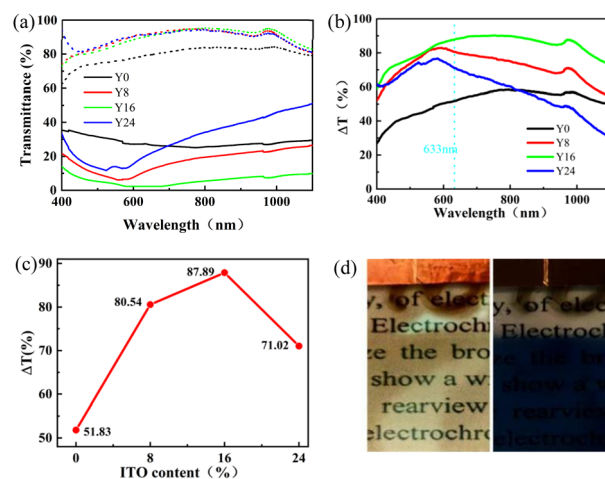


Fig. 4 (a) The optical transmittance spectra of the Y0, Y8, Y16, Y24 films for coloring (solid lines) and for bleaching (dashed lines). (b) Optical modulation  $\Delta T$  at 633 nm wavelength for the samples Y0–Y24. (c) The variation of  $\Delta T$  versus the ITO contents. (d) Photograph of the bleaching (left) and coloration (right) states for sample Y16.

**Table 1** Optical modulation and responding time for all four samples.  $T_c$  and  $T_b$  represent the transmittance in the colored and bleached states at 633 nm. The maximum transmittance for the corresponding states is denoted by  $\text{Max}T_b/\text{Max}T_c$ . I and II represent the first cycle and 50th cycle, respectively

Samples	$T_b$ (%)	$T_c$ (%)	$\Delta T^I$ (%)	$\Delta T^{II}$ (%)	$\text{Max}T_b$ (%)	$\text{Max}T_c$ (%)	$\text{Max}\Delta T$ (%)
Y0	79.23	29.4	51.83	48.57	84.35	25.1	58.44
Y8	91.55	11.01	80.54	73.50	94.91	6.10	82.92
Y16	90.23	2.34	87.89	55.57	95.39	2.27	90.28
Y24	91.16	20.14	71.02	77.17	94.53	11.63	76.69

applications of EC films and devices. The switching speed is defined as the time required for 90% of the full optical transmittance contrast when the EC material is in the colored/bleached state. The calculated results for different samples with ITO content from the measured transmission in Fig. 5(a) are shown in Fig. 5(b), where  $t_b$  and  $t_c$  correspond to the bleaching/coloration time, respectively. A remarkable shortening of the bleaching time is observed in Fig. 5(b) after introducing ITO nanoparticles to the  $\text{WO}_3$  film. Basically, the coloration times  $t_c$  of the  $\text{WO}_3/\text{ITO}$  nanocomposite films are longer than the bleaching time  $t_b$ , indicating that the charge/ $\text{Al}^{3+}$  ion insertion process is more difficult than the extraction process. The shortest bleaching and coloration times are 14.9 s and 16.1 s, respectively, corresponding to 8% content of ITO nanoparticles at the composite film. As the content of ITO is 24%, the bleaching time  $t_b$  is 12.9 seconds, which exceeds the reported responding time of the  $\text{WO}_3$  nanosheet.<sup>32,33</sup> Therefore, the addition of ITO nanoparticles into the monoclinic  $\text{WO}_3$  nanosheet is an effective way to improve its switching speed. The underlying physical mechanism could be attributed to the increased conductivity of the  $\text{WO}_3/\text{ITO}$  film due to the introduction of ITO nanoparticles.

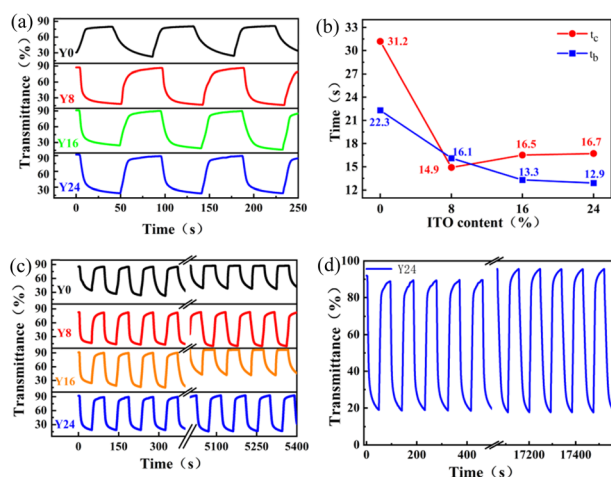
Next, we focus on the cycling stability of the as-prepared samples. Fig. 5(c) shows the transmission spectra of all of the

samples at 633 nm at an applied voltage from  $-0.7$  V to  $+0.7$  V for 50 cycles. It can be seen that the samples illustrate higher transmittance at the coloration state with the increase of the cycling times because a small amount of the  $\text{WO}_3$  powder drops from the glass substrate in the process of measurement. The most stable transmittance sample is Y24 (24% content of ITO), which exhibits almost the same transmittance at the coloration/bleaching state even after 180 cycles (Fig. 5(d)). This phenomenon shows that the film becomes a more stable film with a higher content of ITO in the samples.

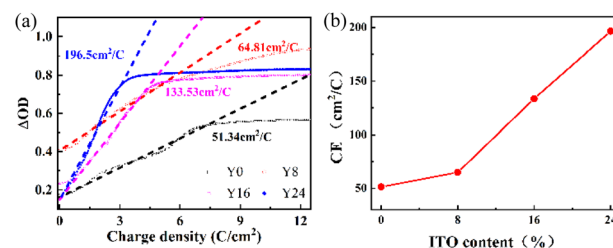
Finally, we discuss the coloration efficiency (CE) for the prepared samples, which is defined as the ratio of the optical density ( $\Delta\text{OD}$ ) variation at a certain wavelength and the corresponding inserted (or extracted) charge density ( $Q$ ) per unit area of the film. CE denotes the ability of the optical modulation at a certain charge density for a designed EC device. Here, CE is calculated by the following equation:

$$\text{CE} = \frac{\Delta\text{OD}}{Q} = \frac{\log(T_b/T_c)}{Q} \quad (2)$$

where  $T_b$  and  $T_c$  are the transmittance of the composite film in the bleaching and coloration states at a certain wavelength, respectively. The charge density is calculated according to eqn (2) by the measured current–time curves for all  $\text{WO}_3$  composite film samples under potential values of  $-0.7$  V and  $+0.7$  V (633 nm wavelength). The optical density  $\Delta\text{OD}$  values at 633 nm as a function of the inserted charge density at a potential of  $-0.7$  V are shown in Fig. 6(a). The value of CE (dashed) is denoted by the slope of the solid curves, and it is improved significantly with the increase of the ITO contents from 0 to 24%. The variation tendency of CE with ITO content is displayed in Fig. 6(b). The measured CE value is much larger than the previously published results for the  $\text{WO}_3$  nanocrystal without



**Fig. 5** (a) The transmittance spectra for samples Y0, Y8, Y16, Y24. (b) Bleaching and coloration times as a function of the ITO content. (c) Electrochromic switching behaviors of all samples at 630 nm at an applied voltage from  $-0.7$  V to  $+0.7$  V for 50 cycles. (d) Transmittance for sample Y24 within 180 cycles.



**Fig. 6** (a) Optical density (solid) of Y0, Y8, Y16, Y24 versus the charge density at wavelength 633 nm, and the slope denoted by the dashed line is the value of CE. (b) CE variation with the ITO content.





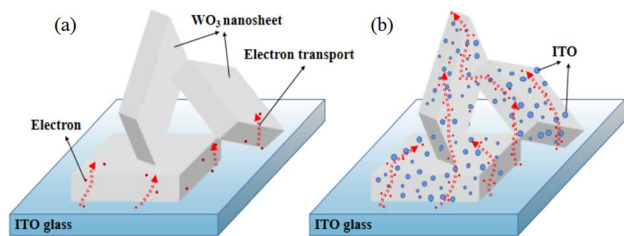


Fig. 7 Schematic illustration of the transportation of electrons through a pure  $\text{WO}_3$  film (a) and a  $\text{WO}_3$ /ITO composite film (b). The red and blue dots denote electrons and ITO nanoparticles, respectively.

the doping of ITO.<sup>12,32,33</sup> This property also can be ascribed to the improved electrochemical activity and conductivity with the combined  $\text{WO}_3$  sheet nanostructure and ITO nanoparticles. The increased conductivity of the  $\text{WO}_3$ /ITO composite film shortens the  $\text{Al}^{3+}$  diffusion pathway, promoting the transmittance modulation with a small intercalation charge density.

To explain the underlying mechanism of the enhanced EC properties discussed above, Fig. 7 displays a schematic of the electron transport process. For a pure  $\text{WO}_3$  film (without ITO nanoparticles),  $\text{WO}_3$  possesses a high charge-transport-barrier because of its poor electrical conductivity,<sup>23</sup> and electrons can only transport from the contact point of the  $\text{WO}_3$  nanosheets and substrate, as shown in Fig. 7(a). As for the  $\text{WO}_3$ /ITO composite film illustrated in Fig. 7(b), conductive ITO nanoparticles attached on the  $\text{WO}_3$  nanosheets serve as many extended 3-dimensional electrodes. Therefore, more  $\text{WO}_3$  nanoparticles could participate in the EC reaction, which increases the electron exchange area and accelerates the EC reaction speed. In addition, for the reported one-dimensional  $\text{WO}_3$  nanostructures such as nanorods, the electrons mainly accumulate at the contact points of the  $\text{WO}_3$  nanorods and the ITO glass substrate. However, for our prepared  $\text{WO}_3$ /ITO nanosheets composite film, the ITO nanoparticles are uniformly distributed around the  $\text{WO}_3$  nanosheets, leading to a more homogeneous EC reaction.

## 4. Conclusion

In summary, we have demonstrated enhanced electrochromic properties through introducing ITO nanoparticles into the monoclinic  $\text{WO}_3$  nanosheet and ITO composite film. The optical modulation has improved to 87.89% at the wavelength of 633 nm, which is much larger than the values of most reported pristine  $\text{WO}_3$  films. The switching speed and cycling stability are also enhanced by adding ITO nanoparticles into the  $\text{WO}_3$  nanosheet. Particularly, the coloration efficiency is boosted remarkably in the prepared composite EC film. These enhanced EC properties are explained by the increased conductivity and electrochemical activity because the coated composite film forms an effective conductive network on the ITO substrate. The excellent EC properties and low-cost fabrication method of the composite  $\text{WO}_3$ /ITO films suggest its broad applications in commercial production of electrochromic smart devices and great potential for other electrochemical electronics.

## Author contributions

Yuan Yu Zheng performed the experiments and characterization. Feng Hui An and Jian Qiang Liu analyzed the data and wrote the manuscript. Meng Dong He and Bo Zhang provided helpful discussions and revised the manuscript. The first two authors contributed to this work equally. All authors have given approval to the final version of the manuscript.

## Conflicts of interest

There are no conflicts to declare.

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