


 Cite this: *RSC Adv.*, 2023, **13**, 12731

Metal organic framework-loaded polyethersulfone/polyacrylonitrile photocatalytic nanofibrous membranes under visible light irradiation for the removal of Cr(vi) and phenol from water

Shahnaz Koushkbaghi,^a Hamta Arjmand Kermani,^a Sana Jamshidifard,^b Hamed Faramarzi,^c Mina Khosravi,^d Parvaneh Ghaderi-shekhi Abadi,^e Fariborz Sharifian Jazi^f and Mohammad Irani^g    

In this work, various amounts of the UiO-66-NH₂ and UiO-66-NH₂/TiO₂ MOFs have been loaded into polyacrylonitrile (PAN) nanofibers supported on polyethersulfone (PES). The visible light irradiation was used to investigate the influence of pH (2–10), initial concentration (10–500 mg L⁻¹), and time (5–240 min) on the removal efficiency of phenol and Cr(vi) in the presence of MOFs. The reaction time: 120 min, catalyst dosage: 0.5 g L⁻¹, pH: 2 for Cr(vi) ions and pH: 3 for phenol molecules were optimum to degrade phenol and to reduce Cr(vi) ions. The characterization of the produced samples was performed using X-ray diffraction, ultraviolet-visible diffuse reflectance spectroscopy, scanning electron microscopy, and Brunauer–Emmett–Teller analysis. The capability of synthesized photocatalytic membranes was investigated for the removal of phenol and Cr(vi) ions from water. The water flux, Cr(vi) and phenol solutions fluxes and their rejection percentages were evaluated under pressure of 2 bar in the presence of visible light irradiation and in the dark. The best performance of the synthesized nanofibers was obtained for UiO-66-NH₂/TiO₂ MOF 5 wt% loaded-PES/PAN nanofibrous membranes at temperature of 25 °C and pH of 3. Results demonstrated the high capability of MOFs-loaded nanofibrous membranes for the removal of various contaminants such as Cr(vi) ions and phenol molecules from water.

Received 12th February 2023
 Accepted 31st March 2023

DOI: 10.1039/d3ra00959a
rsc.li/rsc-advances

1. Introduction

The rapid development of industry and shortage of water resources caused the development of novel alternative faster methods for the rapid removal of contaminants from water.¹ Various technologies including advanced oxidation processes (AOPs), membrane separation, coagulation, adsorption, and ion exchange have been used to remove toxic matters from water.² Recently, hybrid methods such as adsorption/photocatalysis,^{3–5} coagulation/adsorption^{6,7} and photocatalysis/membrane^{8–11} have been developed to increase the removal efficiency of

effluents and accelerate their treatment compared with simple treatment techniques. The photocatalysis/membrane technique is a physical separation/chemical oxidation combined method for the reduction of membrane fouling and increasing the removal efficiency of membranes.¹⁰

Metal organic frameworks (MOFs) as novel photocatalysts have been utilized for degrading organic effluents and reducing metal ions, due to their adjustable pores, high surface area, and high photocatalytic activity through the charge transfer between organic ligand–metal cluster under visible light irradiation.^{12–16} MOFs used for photo-degradation of toxic matters from aquatic systems include various types of UiO, MIL, and ZIF.¹⁷ However, the use of pure MOFs due to difficult recycling after the photocatalysis process is limited.^{18–20} The MOFs loaded membranes and development of photocatalytic membranes is an effective method for (I) uniform disposition of MOFs on the support, (II) use of MOFs in large-scale experiments, (III) prevention of their agglomeration during the photocatalysis process, and (IV) easier recycling after removal of effluents.²¹ For instance, Du *et al.*²² investigated the performance of UiO-66-NH₂ membrane supported on α -Al₂O₃ under sunlight irradiation for reduction of Cr(vi) ions. Liu *et al.*²³ incorporated the Ni@UiO-66 MOFs into

^aScience and Research Branch, Islamic Azad University, Yazd, Iran

^bFaculty of Chemical Engineering, Iran University of Science & Technology, Tehran, Iran

^cChemical Engineering Departments, Razi University, Kermanshah, Iran

^dDepartment of Environmental Engineering, Graduate Faculty of Environment, University of Tehran, Tehran, Iran

^eEnvironmental Health Engineering Research Center, Alborz University of Medical Sciences, Karaj, Iran

^fSchool of Science and Technology, The University of Georgia, Tbilisi, Georgia

^gDepartment of Pharmaceutics, Faculty of Pharmacy, Alborz University of Medical Sciences, Karaj, Iran. E-mail: Irani_mo@ut.ac.ir



the polyethersulfone (PES) membrane under UV irradiation for the water treatment. Ahmadi *et al.*²⁴ immobilized 0.2 wt% NH₂-MIL125(Ti) MOF on the polysulfone membrane for photodegradation of methylene blue under UV irradiation. They also suspended the MOF nanoparticles in the reactor. The methylene blue removal efficiency and flux recovery ratio were 97% and 88%, respectively. Sun *et al.*²⁵ incorporated the poly(sulfobetaine methacrylate)/UiO-66 composite into the polysulfone ultrafiltration membrane. The water flux of the MOF-based composite-incorporated polysulfone was higher than that of the polysulfone membrane (about 2.5 times). Salehian *et al.*²⁶ investigated the removal efficiency of natural organic matter using a TiO₂@MIL-88A (Fe)-loaded polyacrylonitrile photocatalytic membrane. The humic acid removal efficiency and flux recovery ratio of the membrane were 92.4% and 99.5%, respectively. The nanofibers prepared by electrospinning are good candidates for incorporating MOFs.^{27,28} In recent years, the nanofibrous mats have been extensively utilized as a membrane in ultrafiltration, microfiltration, nanofiltration and forward osmosis membrane processes.^{29–32} However, the use of nanofibers in the continuous wastewater treatments such as membrane processes due to their low mechanical stability is limited. For instance, Khalil *et al.*³³ investigated the potential of PAN/SiO₂–TiO₂–NH₂ composite nanofibers for degradation of acid red 27 and malachite green under visible light. The rapid degradation of acid red 27 and malachite green using nanofibers was occurred during 9 and 25 min, respectively. In another study, the performance of a SiO₂–TiO₂-loaded polyani-line nanofiber membrane was studied to degrade the methyl orange.³⁴ The prepared PAN/Ag–TiO₂ nanofiber membrane indicated the high photocatalytic activity for the complete removal of methylene blue within 1 h.³⁵ Pu *et al.*³⁶ investigated the degradation of ciprofloxacin using a PAN/ZIF-65 MOFs nanofiber membrane. However, there is no study on the removal of phenol and Cr(vi) using polyethersulfone (PES)/PAN/UiO-66-NH₂/TiO₂ MOFs nanofiber membranes. In this work, the synthesized UiO-66-NH₂/TiO₂ MOFs were first loaded into the polyacrylonitrile (PAN) solution. The PAN/MOFs have been electrospun on the PES nanofibrous support to prepare the PES/PAN/MOFs photocatalytic nanofibrous membranes. The capability of synthesized photocatalytic membrane was investigated for the removal of phenol and Cr(vi) ions from water under visible light irradiation.

2. Experimental

2.1 Materials

Polyacrylonitrile (PAN, Mw = 150 kDa, Sigma-Aldrich, USA), polyether sulfone (PES, Mw = 58 kDa, Ultrason E6020P), 2-aminoterephthalic acid (purity \geq 99.9%, Sigma-Aldrich, USA, BDC-NH₂), zirconium chloride (purity 99%, Sigma-Aldrich, USA, ZrCl₄), *N,N*-dimethylformamide (Merck, Germany, DMF), hydrochloric acid (HCl, 37%, Merck, Germany), and Titanium tetrabutoxide (C₁₀H₃₆O₄Ti, purity 97%, Sigma-Aldrich, USA) were used for the preparation of nanofibrous membranes.

2.2 Synthesis of MOFs

UiO-66-NH₂ and TiO₂ nanoparticles were synthesized using hydrothermal and sol-gel methods as described previously.^{19,37} To prepare UiO-66-NH₂/TiO₂ composites, first 50 mg TiO₂ nanoparticles were dispersed in ethanol. Then, 50 mg UiO-66-NH₂ was dispersed in solution under sonication for 30 min. After that, the synthesized hybrid was filtered and washed three times with water and ethanol. Finally, the produced solid was dried at 100 °C overnight.

2.3 Fabrication of PES/PAN/MOFs membrane

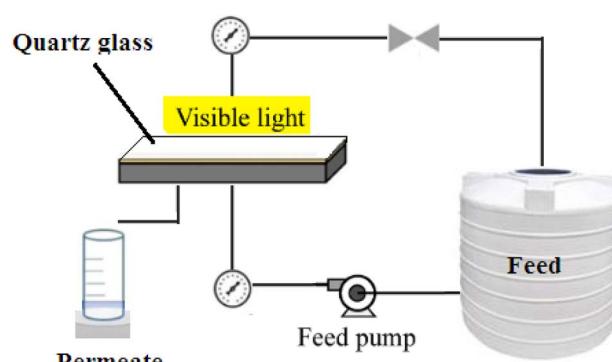
The PES nanofibrous support was prepared by electrospinning method by dissolving 2 g PES in 8 mL DMF and its electrospinning under feeding rate of 1 mL h⁻¹, voltage of 20 kV, and distance of 15 cm. PAN solution was prepared by its dissolving in DMF at 60 °C within 4 h. To prepare the PAN/MOFs and PAN/MOFs/TiO₂ solutions, different amounts of MOFs and MOFs/TiO₂ (2, 5 and 10 wt% by weight of PAN) were dispersed in DMF. Then, PAN was added under stirring overnight. First, the prepared PAN/MOFs and PAN/MOFs/TiO₂ solutions were sonicated for 30 min, and then were electrospun on the PES support.

2.4 Photocatalytic experiments using MOFs

In the photocatalytic removal of phenol and Cr(vi) using MOFs, the impact of initial concentrations of phenol and Cr(vi) (10–500 mg L⁻¹), pH (2–10), and contact time (5–240 min) on their removal using MOFs was investigated under Xenon irradiation (300 W, $\lambda \geq 420$ nm, Beijing Aulight Co., Ltd).

2.5 Photocatalytic membrane experiments

The performance of the PES/PAN/MOFs nanofibrous membranes was examined in a cross-flow photocatalytic membrane reactor under visible light (Xenon arc lamp), operating pressure of 2 bar, effective surface area of 35 cm², and temperature of 25 °C. The filtration was carried out for 120 min with an initial feed concentration of 10 mg L⁻¹. The membranes were regenerated using 0.1 M HCl solution (200 mL) for 2 h.¹⁹ The experimental set-up of the photocatalytic membrane process is illustrated in Scheme 1.



Scheme 1 Experimental set-up of photocatalytic membrane process.



2.6 Characterization tests

The morphology of membranes was detected by employing a scanning electron microscopy (SEM) using, JEOL JSM-6380 microscope. An Image J software (Image-Proplus, Media Cybrernetics) was used to determine the particle size and the size distribution of particles and nanofibers. A diffuse reflectance spectrum (DRS) of MOFs was recorded using UV-2550 (Shimadzu, Japan) UV-vis spectrophotometer. The crystallinity and surface area of synthesized MOFs were determined using X-ray diffractometer type Philips PW 1730 (Japan) and Brunauer–Emmett–Teller (BET) analysis. The contact angle of membranes

was investigated using a contact angle meter (CA-VP, Kyowa Interface Science Co., Ltd, Japan). The pore radius (r_m) of nanofibrous membranes is calculated as follows:

$$r_m = \sqrt{\frac{(2.9 - 1.75\epsilon) \times 8\eta l Q}{\epsilon \times A \times \Delta P}} \quad (1)$$

where η is the water viscosity (8.9×10^{-4} Pa s), Q is the volume of the permeate pure water per unit time ($\text{m}^3 \text{ s}^{-1}$), ΔP is the operating pressure (0.2 MPa), A is the membrane effective area (m^2), l is the thickness of the membrane (m) and ϵ is the porosity of the membrane which is defined as follows:

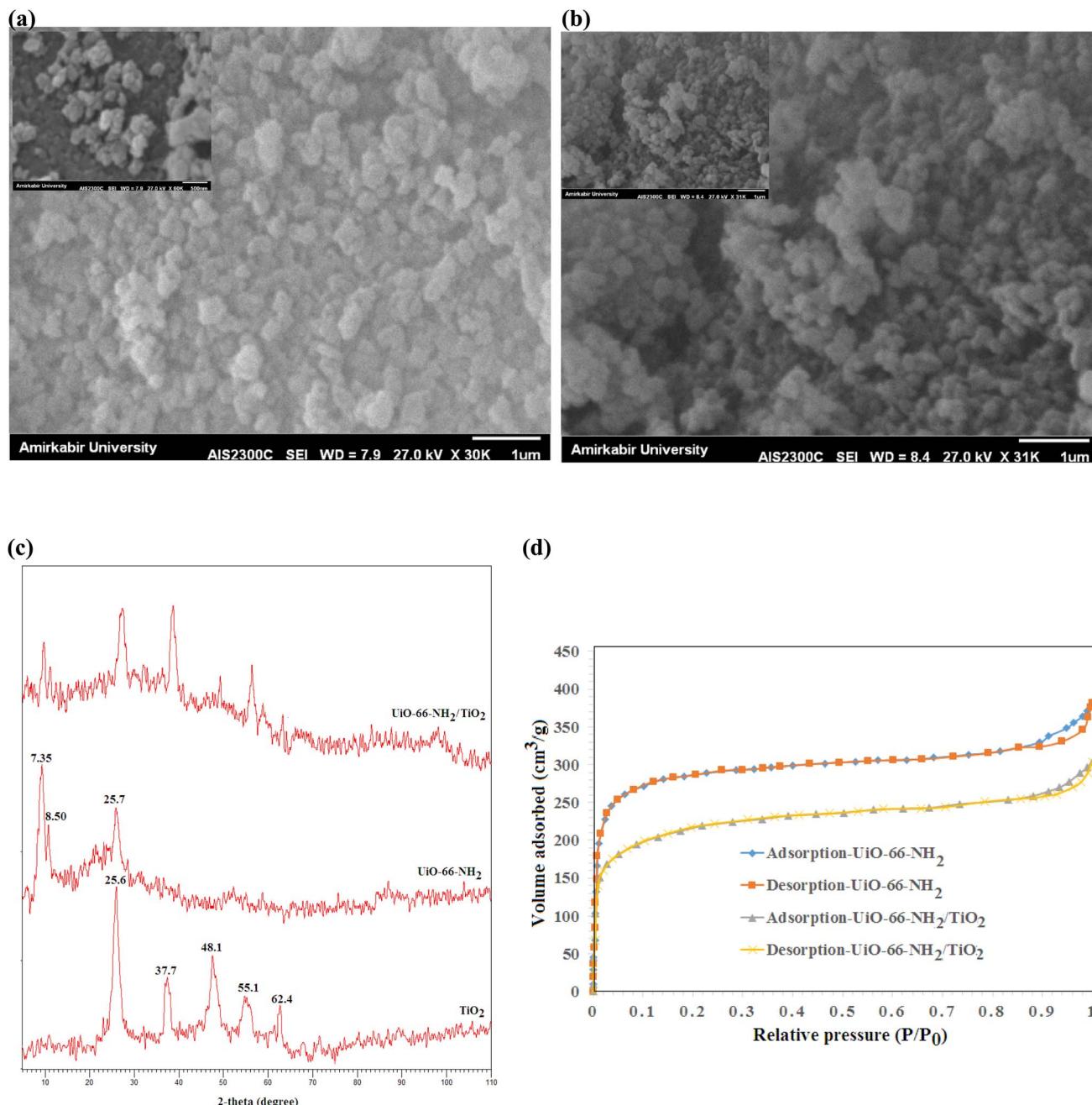


Fig. 1 SEM images of (a) UiO-66-NH_2 (b) $\text{UiO-66-NH}_2/\text{TiO}_2$ composite, (c) XRD patterns of UiO-66-NH_2 and $\text{UiO-66-NH}_2/\text{TiO}_2$ composites and (d) N_2 adsorption/desorption cycles of UiO-66-NH_2 and $\text{UiO-66-NH}_2/\text{TiO}_2$ composites.



$$\varepsilon = \frac{W_1 - W_2}{A \times l \times d_w} \quad (2)$$

where W_1 is the weight of the wet membrane, W_2 is the weight of the dry membrane and d_w is the water density (0.998 g cm^{-3}).

3. Results and discussion

3.1 Characterization of MOFs

The SEM images of UiO-66-NH_2 and $\text{UiO-66-NH}_2/\text{TiO}_2$ composites are illustrated in Fig. 1a and b. The particle sizes

ranging from 150–250 nm with an average size of $185 \pm 45 \text{ nm}$ were obtained for UiO-66-NH_2 MOFs. By blending TiO_2 nanoparticles and UiO-66-NH_2 , the particle sizes ranging from 50–200 nm with an average size of 95 nm have been produced for $\text{UiO-66-NH}_2/\text{TiO}_2$ composite. The XRD patterns of synthesized MOFs are illustrated in Fig. 1c. For UiO-66-NH_2 MOF nanoparticles, the characteristic peaks at 7.35° , 8.50° and 25.7° indicated the successful synthesis of UiO-66-NH_2 .¹⁹ For pure TiO_2 nanoparticles, the detected peaks at 25.6° , 37.7° , 48.1° , 55.1° and 62.4° corresponding to the $(1\ 0\ 1)$, $(0\ 0\ 4)$, $(2\ 0\ 0)$, $(2\ 1$

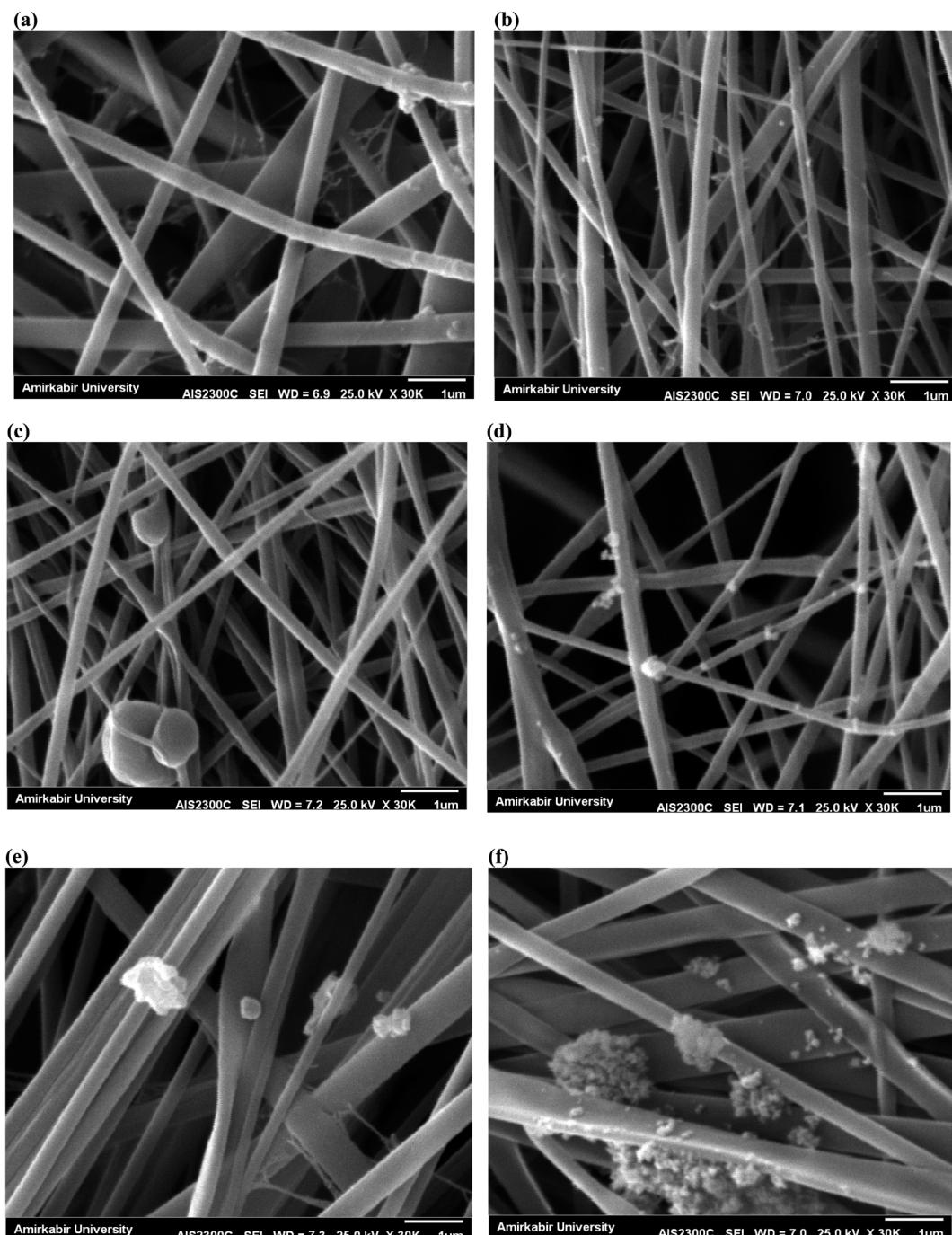


Fig. 2 SEM images of (a) PES, (b) PAN, (c) PAN/ UiO-66-NH_2 5%, (d) PAN/ $\text{UiO-66-NH}_2/\text{TiO}_2$ 2%, (e) PAN/ $\text{UiO-66-NH}_2/\text{TiO}_2$ 5%, and (f) PAN/ $\text{UiO-66-NH}_2/\text{TiO}_2$ 10%.



1) and (2 0 4) lattice planes demonstrated the anatase phase of TiO_2 nanoparticles.³⁷ The main peaks of UiO-66-NH_2 and TiO_2 nanoparticles were matched in the XRD pattern of $\text{UiO-66-NH}_2/\text{TiO}_2$ composite. The N_2 adsorption/desorption cycles in the structure of UiO-66-NH_2 and $\text{UiO-66-NH}_2/\text{TiO}_2$ MOFs are illustrated in Fig. 1d. The surface area of UiO-66-NH_2 and $\text{UiO-66-NH}_2/\text{TiO}_2$ MOFs were found to be 825.2 and $410.1 \text{ m}^2 \text{ g}^{-1}$, respectively. By blending UiO-66-NH_2 and TiO_2 , some TiO_2 nanoparticles were aggregated on the UiO-66-NH_2 surface and decreased the BET surface area and pore volume of UiO-66-NH_2 . Furthermore, the lower surface area of TiO_2 nanoparticles compared to UiO-66-NH_2 MOFs resulted in the lower surface area of $\text{UiO-66-NH}_2/\text{TiO}_2$ in comparison to pure UiO-66-NH_2 .

3.2 Characterization of nanofibrous membranes

The SEM images of the surface of PES nanofibrous support and PAN nanofibrous membranes with different content of MOFs (0, 2, 5 and 10 wt%) are presented in Fig. 2. The homogeneous nanofibers with average diameters of $360 \pm 60 \text{ nm}$ and $250 \pm 50 \text{ nm}$ were obtained for pure PES (Fig. 2a) and PAN (Fig. 2b) nanofibers, respectively. By loading 5 wt% UiO-66-NH_2 , some MOFs were observed on the nanofibers surface and the average diameter of nanofibers was increased to $330 \pm 120 \text{ nm}$ (Fig. 2c). The similar morphology with an average diameter of $315 \pm 100 \text{ nm}$ was obtained for 5 wt% $\text{UiO-66-NH}_2/\text{TiO}_2$ -loaded PAN nanofibers (Fig. 2e). By loading 2 wt% $\text{UiO-66-NH}_2/\text{TiO}_2$ MOFs, the thinner fibers with average diameter of $280 \pm 60 \text{ nm}$ have been prepared and the most of MOFs nanoparticles without aggregation have been successfully incorporated into the nanofibers (Fig. 2d). By loading 2 wt% $\text{UiO-66-NH}_2/\text{TiO}_2$ MOFs into the PAN nanofibers, the viscosity of electrospinning solution was increased which resulted in gradual increase in the fiber diameter of $\text{PAN/UiO-66-NH}_2/\text{TiO}_2$ 2 wt% ($280 \pm 60 \text{ nm}$) compared to pure PAN nanofibers ($250 \pm 50 \text{ nm}$). By increasing the concentration of $\text{UiO-66-NH}_2/\text{TiO}_2$ in the PAN solution, the aggregation of $\text{UiO-66-NH}_2/\text{TiO}_2$ nanoparticles in the solution and non-homogenous dispersion of nanoparticles resulted in the formation of $\text{UiO-66-NH}_2/\text{TiO}_2$ nanoparticles on the surface of the nanofibers. By loading 10 wt% $\text{UiO-66-NH}_2/\text{TiO}_2$ MOFs, most of MOFs were aggregated on the nanofibers surface (Fig. 2f). The electrospinning of PES/PAN/ $\text{UiO-66-NH}_2/\text{TiO}_2$ MOFs with higher $\text{UiO-66-NH}_2/\text{TiO}_2$ concentrations than 10%, due to the high viscosity of solution and aggregation of nanoparticles in the solution before the electrospinning, was impossible. The other structural parameters of synthesized nanofibrous membranes is listed in Table 1. The water contact angle of pure PES/PAN, $\text{PES/PAN/UiO-66-NH}_2/\text{TiO}_2$ 2%, $\text{PES/$

$\text{UiO-66-NH}_2/\text{TiO}_2$ 5% and $\text{PES/PAN/UiO-66-NH}_2/\text{TiO}_2$ 10% nanofibrous membranes were found to be $77.3 \pm 1.2^\circ$, $65.6 \pm 1.4^\circ$, $49.8 \pm 1.3^\circ$, and $38.8 \pm 1.2^\circ$, respectively. The average thickness of nanofibrous membranes was about $75 \pm 5 \mu\text{m}$. The average pore size and porosity of pure PES/PAN nanofibers were $2.98 \mu\text{m}$, and 72.3%. By loading $\text{UiO-66-NH}_2/\text{TiO}_2$ up to 5% into the nanofibrous membrane, the porosity and pore size of nanofibers was gradually increased and a further increase in the $\text{UiO-66-NH}_2/\text{TiO}_2$ content (10 wt%) resulted in decreasing the porosity and pore size of nanofibrous membranes. The increase in the porosity of nanofibers by loading of $\text{UiO-66-NH}_2/\text{TiO}_2$ could be attributed to the higher porosity of $\text{UiO-66-NH}_2/\text{TiO}_2$ in the nanofibers. The decrease in the porosity and pore size of nanofibers containing 10 wt% $\text{UiO-66-NH}_2/\text{TiO}_2$ could be attributed to the nanoparticles aggregation and there are not enough free voids to equilaterally distribute the nanoparticles into the nanofibers, as confirmed by SEM image.

3.3 Photocatalytic removal of Cr(vi) and phenol in a batch system

The UV diffuse reflectance spectra (DRS) of synthesized MOFs are illustrated in Fig. 3. As shown, the absorption edge of TiO_2 , UiO-66-NH_2 , and $\text{UiO-66-NH}_2/\text{TiO}_2$ MOFs were found to be 388.6 nm , 435.1 nm , and 421.8 nm respectively, indicating that UiO-66-NH_2 , and $\text{UiO-66-NH}_2/\text{TiO}_2$ MOFs could be activated under visible light irradiation. The band-gap energy of UiO-66-NH_2 , and $\text{UiO-66-NH}_2/\text{TiO}_2$ MOFs was estimated to be 2.85 eV and 2.94 eV , respectively.

The effect of pH on the photo-degradation of phenol and Cr(vi) using MOFs under visible light, catalyst dosage of 0.5 g L^{-1} , initial concentration of 10 mg L^{-1} , reaction time of 240 min, temperature of 25°C , and pH values ranging from 2–10 is illustrated in Fig. 3b. As shown, the maximum removal of Cr(vi) using UiO-66-NH_2 , and $\text{UiO-66-NH}_2/\text{TiO}_2$ MOFs was occurred at pH 2. At lower pH values, the better reduction of $\text{Cr}_2\text{O}_7^{2-}$ ions was occurred, due to the better electrostatic attraction of Cr(vi) anions and synthesized MOFs. After that, the removal of Cr(vi) ions was occurred by irradiation of visible light on the MOFs surface *via* the photogenerated-electron-hole pairs (eqn (3) and (4)). At higher pH values, the precipitation of chromium anions in the form of Cr(OH)_3 might cover the active sites of synthesized photocatalysts and reduced their photocatalytic efficiency (eqn (5)).

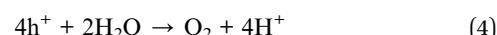
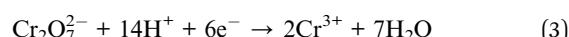


Table 1 Structural parameters of the fabricated nanofibrous membranes

Membrane	Water contact angle (°)	Pore size (μm)	Porosity (%)
PES/PAN	77.3 ± 1.2	2.98 ± 0.15	72.3 ± 1.3
PES/PAN/ $\text{UiO-66-NH}_2/\text{TiO}_2$ 2%	65.6 ± 1.4	3.85 ± 0.13	76.7 ± 0.9
PES/PAN/ $\text{UiO-66-NH}_2/\text{TiO}_2$ 5%	49.8 ± 1.3	5.25 ± 0.21	81.3 ± 1.1
PES/PAN/ $\text{UiO-66-NH}_2/\text{TiO}_2$ 10%	38.8 ± 1.2	1.96 ± 0.10	78.5 ± 1.9



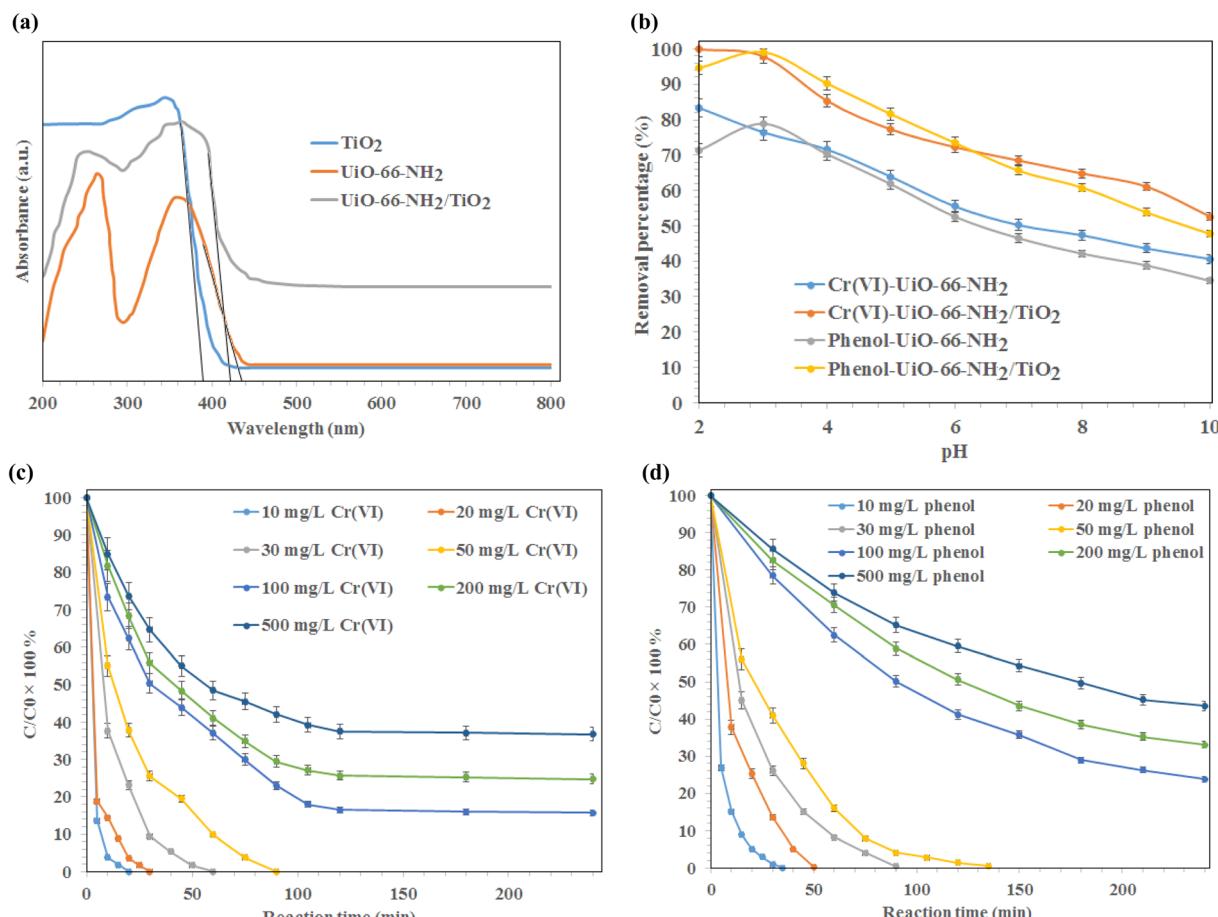


Fig. 3 (a) UV-DRS spectra of synthesized TiO₂, UiO-66-NH₂, and UiO-66-NH₂/TiO₂ MOFs, (b) effect of pH on the photo-degradation of Cr(VI) and phenol using UiO-66-NH₂, and UiO-66-NH₂/TiO₂ MOFs under visible light irradiation, and the effect of reaction time on the removal of (c) Cr(VI) and (d) phenol using UiO-66-NH₂/TiO₂.



The optimum pH for the removal of phenol using synthesized photocatalysts was occurred at pH 3. As shown, the complete degradation of phenol was obtained using UiO-66-NH₂/TiO₂ MOFs at pH 3 after 120 min. The maximum phenol removal percentages in the presence UiO-66-NH₂, and UiO-66-NH₂/TiO₂ MOFs were 81.3% and 99.5%, respectively. Therefore, the pH values of 2 and 3 were selected for further experiments.

The effect of reaction time on the removal of Cr(VI) ions and phenol at various concentrations (10–500 mg L⁻¹) using UiO-66-NH₂/TiO₂ MOFs is illustrated in Fig. 3c and d. As shown, the complete reduction of chromium ions by UiO-66-NH₂/TiO₂ MOFs for initial concentrations of 10, 20, 30, and 50 mg L⁻¹ was occurred after 20, 30, 60 and 90 min, respectively. The maximum Cr(VI) removal percentages for initial concentrations of 100, 200 and 500 mg L⁻¹ Cr(VI) ions were found to be 83.5% ± 1.5%, 74.3% ± 1.7% and 62.5% ± 2.1%, respectively, after 120 min. The phenol removal percentage higher than 99% was obtained using UiO-66-NH₂/TiO₂ MOFs under initial concentrations of 10, 20, 30, and 50 mg L⁻¹ after 35 min, 50 min, 90 min and 135 min, respectively (Fig. 3d). The maximum phenol degradation was found to be 76.2% ± 1.3%, 66.9% ±

1.8% and 56.5% ± 1.9%, respectively, after 240 min for initial concentrations of 100, 200 and 500 mg L⁻¹ phenol. The higher removal percentages of Cr(VI) and phenol using UiO-66-NH₂/TiO₂ MOFs than the UiO-66-NH₂ could be attributed to the higher photocatalytic activity of UiO-66-NH₂/TiO₂ composite. Although, the pure UiO-66-NH₂ MOFs exhibited the higher specific surface area and lower band-gap energy compared to UiO-66-NH₂/TiO₂ composite, the contact interfaces between TiO₂ and UiO-66 promoted the separation/migration efficiency of photogenerated electron/hole pairs during photocatalytic reaction and resulted in increasing the photocatalytic activity of UiO-66-NH₂ for the removal of Cr(VI) and phenol from water.³⁸

3.4 Photocatalytic membranes

The water permeation, Cr(VI) solution flux and phenol solution flux were evaluated at the pressure of 2 bar under visible light irradiation and without light irradiation (Fig. 4). As shown in Fig. 4a, the permeability of the PES/PAN nanofibrous membrane was increased by incorporating UiO-66-NH₂/TiO₂ MOFs into the PES/PAN membrane. The water permeability of PES/PAN, PES/PAN/UiO-66-NH₂/TiO₂ 2%, PES/PAN/UiO-66-NH₂/TiO₂ 5% and PES/PAN/UiO-66-NH₂/TiO₂ 10% nanofibrous



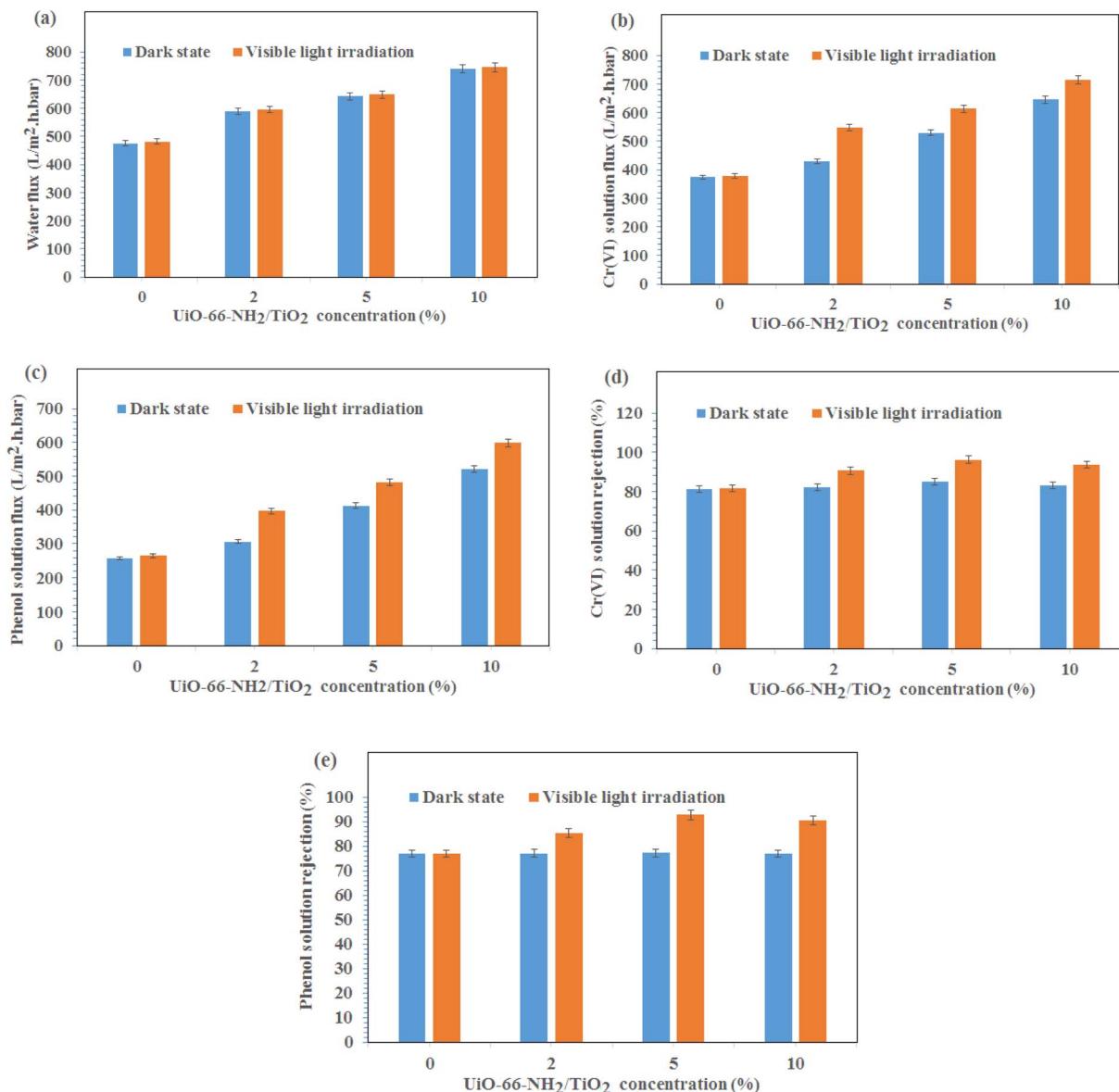


Fig. 4 (a) Water permeation, (b) Cr(vi) solution flux, (c) phenol solution flux, (d) Cr(vi) rejection, (e) phenol rejection at the pressure of 2 bar under visible light irradiation and dark state.

membranes was found to be $475.2 \text{ L m}^{-2} \text{ h}^{-1} \text{ bar}^{-1}$, $589.4 \text{ L m}^{-2} \text{ h}^{-1} \text{ bar}^{-1}$, $643.6 \text{ L m}^{-2} \text{ h}^{-1} \text{ bar}^{-1}$, and $739.3 \text{ L m}^{-2} \text{ h}^{-1} \text{ bar}^{-1}$, respectively. By increasing the concentration of $\text{UiO-66-NH}_2/\text{TiO}_2$, the hydrophilicity of membrane was increased which resulted in increasing the water permeability. The water contact angle of pure PES/PAN, PES/PAN/ $\text{UiO-66-NH}_2/\text{TiO}_2$ 2%, PES/PAN/ $\text{UiO-66-NH}_2/\text{TiO}_2$ 5% and PES/PAN/ $\text{UiO-66-NH}_2/\text{TiO}_2$ 10% nanofibrous membranes were found to be $77.3 \pm 1.2^\circ$, $65.6 \pm 1.4^\circ$, $49.8 \pm 1.3^\circ$, and $38.8 \pm 1.2^\circ$, respectively. The enrichment of the surface of membranes with $-\text{NH}_2$ and Ti-O groups, resulted in decreasing of the water contact angle and increasing the hydrophilicity of membranes by increasing $\text{UiO-66-NH}_2/\text{TiO}_2$ concentration in the PES/PAN membrane. Furthermore, the loading of $\text{UiO-66-NH}_2/\text{TiO}_2$ with high porosity into the membrane may be increased the membrane porosity and

enhanced the water permeability of PES/PAN nanofibrous membrane. The light irradiation did not impact on the water permeability. This behavior indicated no significant reaction between the hydroxyl radicals and polymer chains. Therefore, the intrinsic resistance of the membrane exhibited a critical role on the water permeability. The blocking of some pores of nanofibrous membranes with phenol and Cr(vi) resulted in a gradual decrease of phenol and Cr(vi) solutions compared with the water permeability of PES/PAN/ $\text{UiO-66-NH}_2/\text{TiO}_2$ MOFs nanofibrous membranes (Fig. 4b and c). The Cr(vi) and phenol solutions fluxes have been increased in the presence of visible light irradiation. The photocatalytic degradation of Cr(vi) ions and phenol molecules that blocked the nanofibers pores, resulted in improving the Cr(vi) and phenol solutions permeability under visible light.

The gradual enhancement of Cr(vi) rejection by increasing the concentration of $\text{UiO-66-NH}_2/\text{TiO}_2$ was due to the increasing the hydrophilicity under dark state (Fig. 4d). However, the rejection of phenol did not significantly change by loading of $\text{UiO-66-NH}_2/\text{TiO}_2$ (Fig. 4e). The photodegradation of phenol and Cr(vi) ions by hydroxyl radicals resulted in increasing removal efficiencies of phenol and Cr(vi) under visible light (Fig. 4d and e). The removal efficiencies of phenol and Cr(vi) using PES/PAN/ $\text{UiO-66-NH}_2/\text{TiO}_2$ 5% were 84.9 and 77.3% under dark state. Whereas, the maximum removal efficiencies of phenol and Cr(vi) were 92.7 and 96.3% in the presence of PES/PAN/ $\text{UiO-66-NH}_2/\text{TiO}_2$ 5% nanofibrous membrane under visible light. The gradual decrease in the phenol and Cr(vi) rejection percentages by increasing $\text{UiO-66-NH}_2/\text{TiO}_2$ concentration up to 10 wt% may be attributed to the increase in the membrane porosity and pore radius. Similar trend is reported by Ahmadipouya *et al.*³⁹ They found that the mixed-matrix membrane containing 9 wt% UiO-66 was optimum for the removal of dyes and further loading of UiO-66 MOFs (12 wt%) resulted in decreasing the rejection percentages of dyes.

The phenol solution flux, Cr(vi) solution flux, phenol rejection and Cr(vi) rejection during 120 min in the presence visible light irradiation and without light irradiation are presented in

Fig. 5. The fluxes of phenol and Cr(vi) have decreased from $824.3 \text{ L m}^{-2} \text{ h}^{-1} \text{ bar}^{-1}$ to 529.6 and 633.1 to $412.3 \text{ L m}^{-2} \text{ h}^{-1} \text{ bar}^{-1}$ for phenol and Cr(vi) ions solutions using PES/PAN/ $\text{UiO-66-NH}_2/\text{TiO}_2$ 5% nanofibrous membrane in the dark state. The higher hydrophilicity of nanofibrous membrane containing 5 wt% $\text{UiO-66-NH}_2/\text{TiO}_2$ compared to the hydrophilicity of composite membranes containing lower amounts of $\text{UiO-66-NH}_2/\text{TiO}_2$ resulted in its lower flux decline. At higher amounts of $\text{UiO-66-NH}_2/\text{TiO}_2$, the interaction between contaminants and membrane surface resulted in its gradual higher flux decline compared with 10 wt% $\text{UiO-66-NH}_2/\text{TiO}_2$ loaded- PES/PAN nanofibrous membrane. In the presence visible light, the flux decline has been improved and the minimum flux decline was found to be 26.0% and 25.8% for Cr(vi) and phenol solutions using PES/PAN/ $\text{UiO-66-NH}_2/\text{TiO}_2$ 5% nanofibrous membrane. The hydroxyl radicals generated during photocatalytic reaction could degrade the phenol molecules and Cr(vi) ions and could prevent the flux decline.

The maximum Cr(vi) and phenol rejection percentages were 84.9 and 77.3% under dark state using PES/PAN/ $\text{UiO-66-NH}_2/\text{TiO}_2$ 5% which were due to the adsorption of contaminants by the membrane and a further removal of Cr(vi) and phenol under visible light (phenol: 92.7% and Cr(vi) 96.3%) were due to the

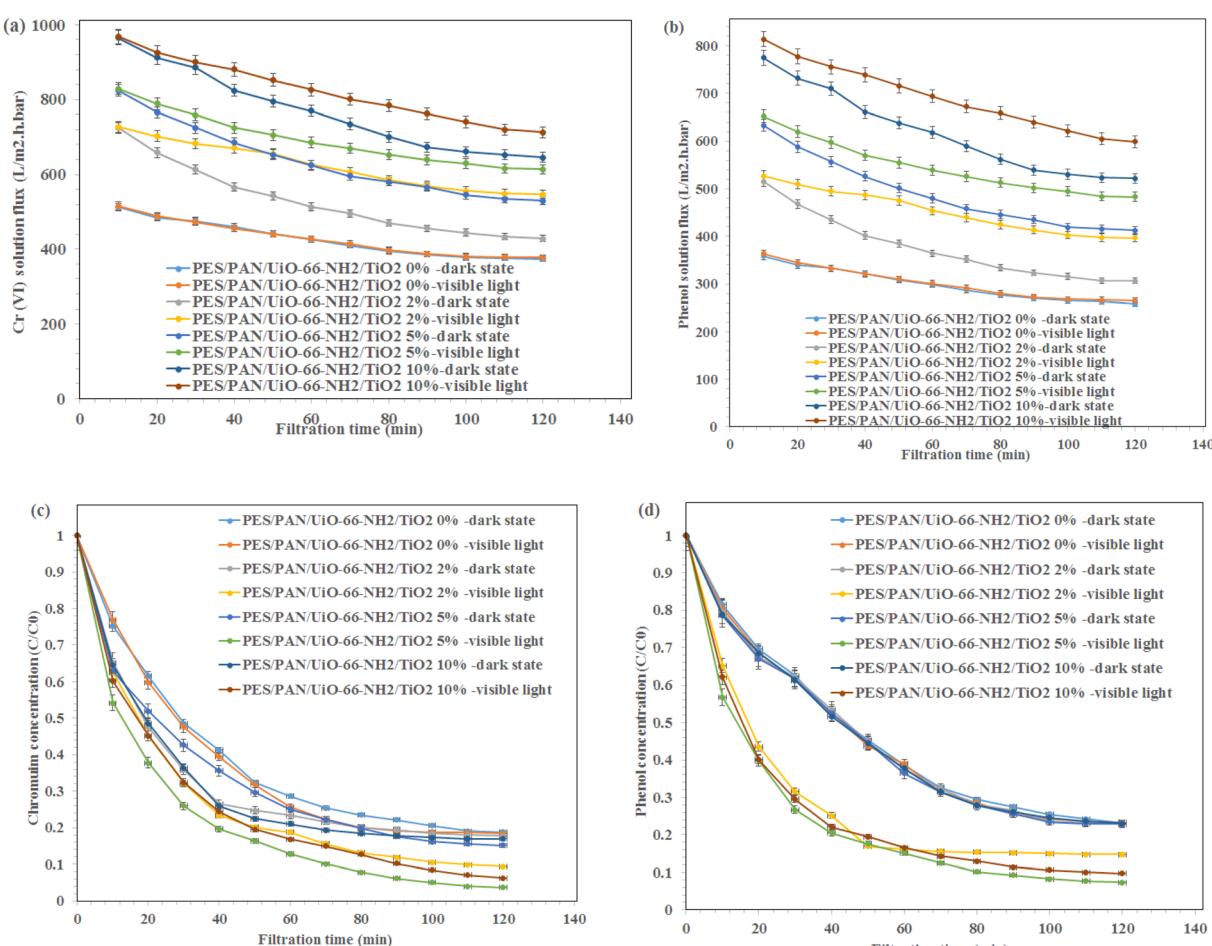


Fig. 5 (a) Cr(vi) and (b) phenol solutions fluxes during 120 min and (c) Cr(vi) and (d) phenol rejection percentages during 120 min in the presence visible light irradiation and dark state.



photocatalytic reduction of contaminants. Therefore, the prepared nanofibrous membranes could eliminate Cr(vi) and phenol from water through the adsorption, filtration, and photocatalytic reduction. For the phenol degradation, the removal efficiency did not significantly change by increasing $\text{UiO-66-NH}_2/\text{TiO}_2$ concentration under the dark state. However, the degradation ability of PES/PAN/ $\text{UiO-66-NH}_2/\text{TiO}_2$ was enhanced by increasing $\text{UiO-66-NH}_2/\text{TiO}_2$ content up to 5%, which due to the enhanced photocatalytic capacity of PES/PAN nanofibers. Therefore, $\text{UiO-66-NH}_2/\text{TiO}_2$ as a photocatalysis composite could improve the performance of PES/PAN/ $\text{UiO-66-NH}_2/\text{TiO}_2$ nanofibrous membrane to degrade the phenol molecules. For Cr(vi) reduction, the removal efficiency of PES/PAN/ $\text{UiO-66-NH}_2/\text{TiO}_2$ was increased by loading $\text{UiO-66-NH}_2/\text{TiO}_2$ into the membrane up to 5% under both dark state and visible light irradiation. Therefore, the adsorption capacity, and photocatalytic reduction of membrane have been improved for

reducing Cr(vi) ions from water. The obtained results indicated that the prepared photocatalytic membrane exhibited a better photocatalytic performance to eliminate Cr(vi) and phenol under visible light irradiation.

The change in the equilibrium fluxes after regeneration of nanofibrous membranes with 0.1 M HCl are illustrated in Fig. 6. As shown, the flux recovery of MOFs-loaded membranes under visible light irradiation was higher than that of the dark state, due to the photocatalytic reactions inside the pores resulting in the enhanced dissolution of the membrane fouling in water, which in turn improved the water flux after cleaning under visible light irradiation.⁴⁰ The equilibrium fluxes of composite nanofibrous membrane containing 5 wt% $\text{UiO-66-NH}_2/\text{TiO}_2$ was maximum before and after rising with HCl under visible light irradiation. This behavior indicated the effect of metal organic framework as a porous material and the photocatalytic reaction on the improvement the performance of the metal organic

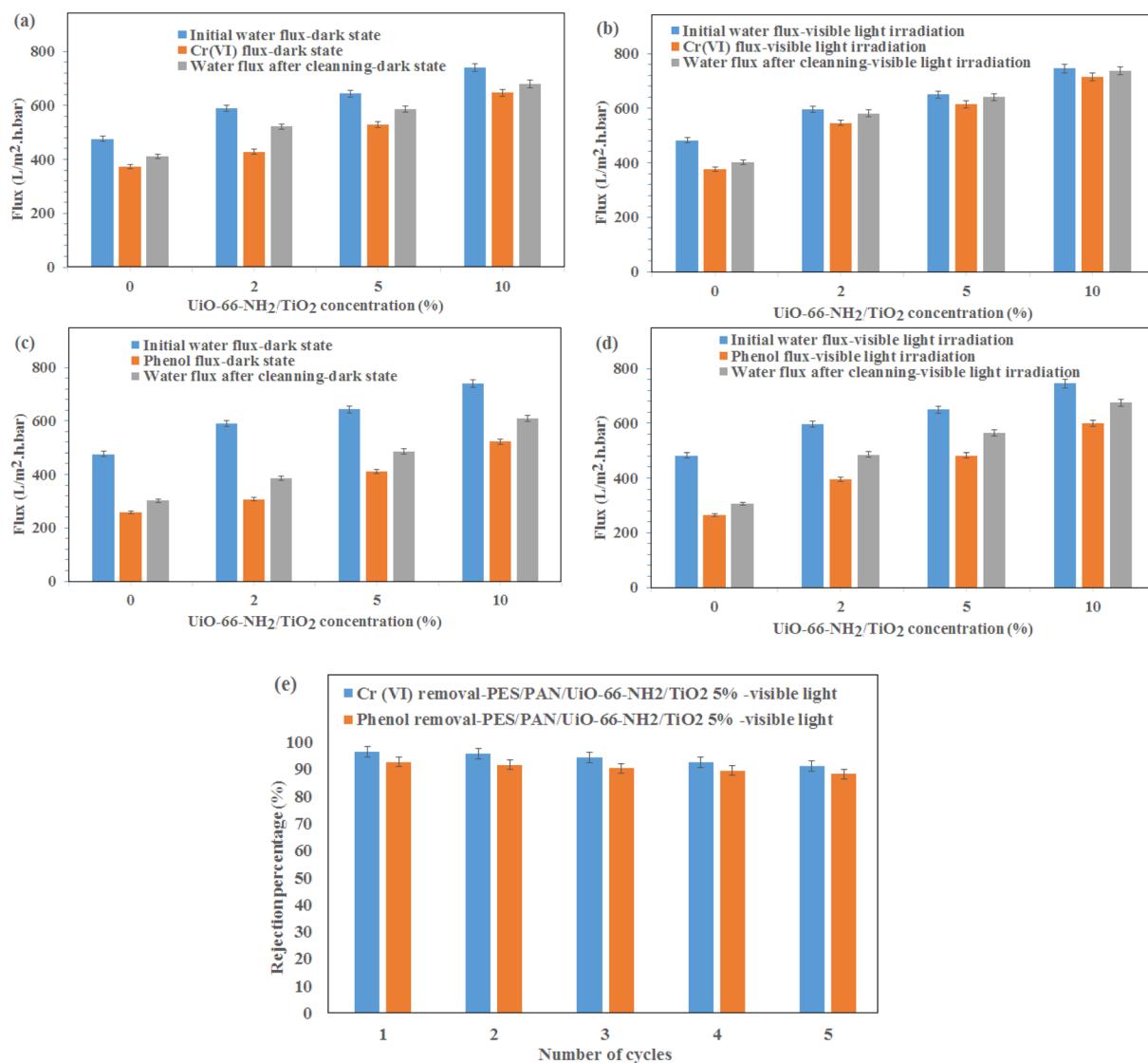


Fig. 6 The change in the equilibrium fluxes of (a) water-Cr(vi) in the dark state, (b) water-Cr(vi) under visible light irradiation, (c) water-phenol in the dark state, (d) water-phenol under visible light irradiation after regeneration of nanofibrous membranes with 0.1 M HCl and (e) Cr(vi) and phenol removal using PES/PAN/ $\text{UiO-66-NH}_2/\text{TiO}_2$ 5% nanofibrous membrane under visible light for five cycles.



framework-based nanofibrous membrane. However, more studies are needed for the reduction of fouling of membranes in the presence photocatalytic reactions.

To investigate the stability of prepared membranes, the Cr(vi) and phenol rejection were investigated for five cycles using PES/PAN/UiO-66-NH₂/TiO₂ 5% nanofibrous membrane under visible light irradiation (Fig. 6e). As shown, the removal efficiencies of Cr(vi) and phenol did not significantly change even after five cycles which demonstrated the stability of the membranes for industrial applications in the future.

4. Conclusion

In this work, the UiO-66-NH₂/TiO₂ MOFs were synthesized via the hydrothermal method. The various amounts of MOFs were incorporated into the PES/PAN nanofibers membranes to investigate the performance of nanofibrous membranes for the removal of Cr(vi) and phenol under visible light irradiation. The UiO-66-NH₂ and UiO-66-NH₂/TiO₂ nanoparticles with average particle size of 185 ± 45 nm and 95 ± 25 nm were produced. The surface area of UiO-66-NH₂ and UiO-66-NH₂/TiO₂ MOFs were found to be 825 and $410 \text{ m}^2 \text{ g}^{-1}$. The average fiber diameter of PAN and PAN nanofibers containing 2,5 and 10 wt% UiO-66-NH₂/TiO₂ MOFs were found to be 250 ± 50 nm, 280 ± 60 nm, 315 ± 100 nm and 410 ± 140 nm, respectively. The maximum Cr(vi) removal percentages for initial concentrations of 100, 200 and 500 mg L⁻¹ were found to be $83.5\% \pm 1.5\%$, $74.3\% \pm 1.7\%$ and $62.5\% \pm 2.1\%$, respectively, after 120 min. The phenol removal percentage higher than 99% was obtained using UiO-66-NH₂/TiO₂ MOFs under initial concentrations of 10, 20, 30, and 50 mg L⁻¹ after 35 min, 50 min, 90 min and 135 min, respectively. The water permeability of PES/PAN, PES/PAN/UiO-66-NH₂/TiO₂ 2%, PES/PAN/UiO-66-NH₂/TiO₂ 5% and PES/PAN/UiO-66-NH₂/TiO₂ 10% nanofibrous membranes was found to be $475.2 \text{ L m}^{-2} \text{ h}^{-1} \text{ bar}^{-1}$, $589.4 \text{ L m}^{-2} \text{ h}^{-1} \text{ bar}^{-1}$, $643.6 \text{ L m}^{-2} \text{ h}^{-1} \text{ bar}^{-1}$, and $739.3 \text{ L m}^{-2} \text{ h}^{-1} \text{ bar}^{-1}$, respectively. The removal efficiencies of phenol and Cr(vi) using PES/PAN/UiO-66-NH₂/TiO₂ 5% were 77.3 and 84.9% under dark state. Whereas, the maximum removal efficiencies of phenol and Cr(vi) were 92.7 and 96.3% in the presence of PES/PAN/UiO-66-NH₂/TiO₂ 5% nanofibrous membrane under visible light irradiation. The equilibrium fluxes of composite nanofibrous membrane containing 5 wt% was maximum before and after rising with HCl under visible light irradiation. The obtained results demonstrated the high capability of MOFs in composite nanofibrous membrane for the removal of various contaminants from water during photocatalytic membrane process.

Conflicts of interest

There are no conflicts to declare.

References

- 1 A. G. Ebadi, M. Toughani, A. Najafi and M. Babaei, *Cent. Asian J. Environ. Sci. Technol. Innov.*, 2020, **1**, 1.
- 2 B. A. Marinho, R. O. Cristóvão, R. A. Boaventura and V. J. Vilar, *Environ. Sci. Pollut. Res.*, 2019, **26**, 2203.
- 3 M. Irani, L. R. Rad, H. Pourahmad and I. Haririan, *Microporous Mesoporous Mater.*, 2015, **206**, 1.
- 4 S. Natarajan, H. C. Bajaj and R. J. Tayade, *J. Environ. Sci.*, 2018, **65**, 201.
- 5 Z. Li, L. Wang, L. Qin, C. Lai, Z. Wang, M. Zhou, L. Xiao, S. Liu and M. Zhang, *Chemosphere*, 2021, **285**, 131432.
- 6 M. Elma, A. E. Pratiwi, A. Rahma, E. L. A. Rampun, M. Mahmud, C. Abdi, R. Rosadi, D. H. Y. Yanto and M. R. Bilad, *Sustainability*, 2021, **14**, 370.
- 7 J. Y. Lin, N. N. N. Mahasti and Y. H. Huang, *J. Hazard. Mater.*, 2021, **407**, 124401.
- 8 S. Mozia, *Sep. Purif. Technol.*, 2010, **73**, 71.
- 9 N. Nasrollahi, L. Ghalamchi, V. Vatanpour and A. Khataee, *J. Ind. Eng. Chem.*, 2021, **93**, 101.
- 10 W. S. Koe, J. W. Lee, W. C. Chong, Y. L. Pang and L. C. Sim, *Environ. Sci. Pollut. Res.*, 2020, **27**, 2522.
- 11 Y. Shi, J. Huang, G. Zeng, W. Cheng and J. Hu, *J. Memb. Sci.*, 2019, **584**, 364.
- 12 A. Dhakshinamoorthy, Z. Li and H. Garcia, *Chem. Soc. Rev.*, 2018, **47**, 8134.
- 13 S. Gautam, H. Agrawal, M. Thakur, A. Akbari, H. Sharda, R. Kaur and M. Amini, *J. Environ. Chem. Eng.*, 2020, **8**, 103726.
- 14 C. C. Wang, J. R. Li, X. L. Lv, Y. Q. Zhang and G. Guo, *Energy Environ. Sci.*, 2014, **7**, 2831.
- 15 X. Zhang, J. Wang, X. X. Dong and Y. K. Lv, *Chemosphere*, 2020, **242**, 125144.
- 16 M. Pishnamazi, S. Koushkbaghi, S. S. Hosseini, M. Darabi, A. Yousefi and M. Irani, *J. Mol. Liq.*, 2020, **317**, 113934.
- 17 S. Zhang, J. Wang, Y. Zhang, J. Ma, L. Huang, S. Yu, L. Chen, G. Song, M. Qiu and X. Wang, *Environ. Pollut.*, 2021, **291**, 118076.
- 18 E. Bahmani, S. Koushkbaghi, M. Darabi, A. ZabihiSahebi, A. Askari and M. Irani, *Carbohydr. Polym.*, 2019, **224**, 115148.
- 19 S. Jamshidifard, S. Koushkbaghi, S. Hosseini, S. Rezaei, A. Karamipour and M. Irani, *J. Hazard. Mater.*, 2019, **368**, 10–20.
- 20 A. Sotto, G. Orcajo, J. M. Arsuaga, G. Calleja and J. Landaburu-Aguirre, *J. Appl. Polym. Sci.*, 2015, **132**, 132.
- 21 H. N. Abdelhamid and A. P. Mathew, *Coord. Chem. Rev.*, 2022, **451**, 214263.
- 22 X. D. Du, X. H. Yi, P. Wang, W. Zheng, J. Deng and C. C. Wang, *Chem. Eng. J.*, 2019, **356**, 393.
- 23 J. Liu, L. Shen, H. Lin, Z. Huang, H. Hong and C. Chen, *J. Colloid Interface Sci.*, 2022, **618**, 483.
- 24 A. Ahmadi, M. H. Sarrafzadeh, A. Hosseini and S. B. Ghaffari, *J. Environ. Chem. Eng.*, 2022, **10**, 106999.
- 25 H. Sun, B. Tang and P. Wu, *ACS Appl. Mater. Interfaces*, 2017, **9**, 21473.
- 26 S. Salehian, M. H. Mehdipour, F. Fotovat and S. A. Mousavi, *Chemosphere*, 2022, **302**, 134893.
- 27 J. E. Efome, D. Rana, T. Matsuura and C. Q. Lan, *J. Mater. Chem. A*, 2018, **6**, 4550.
- 28 H. I. Adil, M. R. Thalji, S. A. Yasin, I. A. Saeed, M. A. Assiri, K. F. Chong and G. A. M. Ali, *Rsc Adv.*, 2022, **12**, 1433.

29 H. Chen, M. Huang, Y. Liu, L. Meng and M. Ma, *Sci. Total Environ.*, 2020, **739**, 139944.

30 C. Feng, K. C. Khulbe, T. Matsuura, S. Tabe and A. F. Ismail, *Sep. Purif. Technol.*, 2013, **102**, 118–135.

31 H. Hadi Najafabadi, M. Irani, L. R. Rad, A. H. Haratameh and I. Haririan, *RSC Adv.*, 2015, **5**, 16532.

32 Y. Liao, C. H. Loh, M. Tian, R. Wang and A. G. Fane, *Prog. Polym. Sci.*, 2018, **77**, 69.

33 A. Khalil, N. M. Aboamera, W. S. Nasser, W. H. Mahmoud and G. G. Mohamed, *Sep. Purif. Technol.*, 2019, **224**, 509.

34 Z. Liu, Y. E. Miao, M. Liu, Q. Ding, W. W. Tjiu, X. Cui and T. Liu, *J. Colloid Interface Sci.*, 2014, **424**, 49.

35 Y. Shi, D. Yang, Y. Li, J. Qu and Z. Z. Yu, *Appl. Surf. Sci.*, 2017, **426**, 622.

36 M. Pu, D. Ye, J. Wan, B. Xu, W. Sun and W. Li, *Sep. Purif. Technol.*, 2022, **299**, 121716.

37 T. Tański, W. Matysiak and L. Krzemiński, *Mater. Manuf. Process.*, 2017, **32**, 1218.

38 Z. Man, Y. Meng, X. Lin, X. Dai, L. Wang and D. Liu, *Chem. Eng. J.*, 2022, **431**, 133952.

39 S. Ahmadipouya, S. A. Mousavi, A. Shokrgozar and D. V. Mousavi, *J. Environ. Chem. Eng.*, 2022, **10**, 107535.

40 M. Kumar, Z. Gholamvand, A. Morrissey, K. Nolan, M. Ulbricht and J. Lawler, *J. Memb. Sci.*, 2016, **506**, 38.

