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# A glucose-assisted redox hydrothermal route to prepare a Mn-doped CeO<sub>2</sub> catalyst for the total catalytic oxidation of VOCs†

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In this study, a novel glucose-assisted redox hydrothermal method has been presented to prepare an Mndoped  $CeO_2$  catalyst (denoted as Mn- $CeO_2$ -R) for the first time. The obtained catalyst contains uniform nanoparticles with a small crystallite size, a large mesopore volume, and rich active surface oxygen species. Such features collectively contribute to improving the catalytic activity for the total catalytic oxidation of methanol ( $CH_3OH$ ) and formaldehyde (HCHO). Interestingly, the large mesopore volume feature of the Mn- $CeO_2$ -R samples could be considered an essential factor to eliminate the diffusion limit, favoring the total oxidation of toluene ( $C_7H_8$ ) at high conversion. Therefore, the Mn- $CeO_2$ -R catalyst outperforms both bare  $CeO_2$  and conventional Mn- $CeO_2$  catalysts with  $T_{90}$  values of 150 °C and 178 °C for HCHO and  $CH_3OH$ , respectively, and 315 °C for  $C_7H_8$ , at a high GHSV of 60 000 mL  $g^{-1}$  h<sup>-1</sup>. Such robust catalytic activities signify a potential utilization of Mn- $CeO_2$ -R for the catalytic oxidation of volatile organic compounds ( $VOC_5$ ).

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## Introduction

Volatile organic compounds (VOCs) emitted from industrial activities have caused concern as the root cause of well-known serious human health and environmental issues.1 Indeed, methanol (CH<sub>3</sub>OH), formaldehyde (HCHO), and toluene (C<sub>7</sub>H<sub>8</sub>), the three most typical harmful VOC pollutants, cause various health problems (e.g., headache, nausea, irritation of the skin, nose, eyes, and throat).2,3 In addition, HCHO molecules stimulate discomfort at low levels and are classified as a group 1 carcinogen by The International Agency for Research on Cancer (IARC). Therefore, catalytic oxidation-driven VOC-to-CO<sub>2</sub> conversion has arisen as an efficient platform to completely remove those mentioned pollutants.4 To this end, noble metalbased catalysts (e.g., Pt, Au, Pd) have been proven effective with high performance.5,6 However, high cost and water vapor density restrict the utilization of precious metals for the catalytic oxidation of VOCs.

Recent attempts have pointed out the enormous potential of cerium oxide  $(CeO_x)$ -based materials as robust total oxidation

catalysts for the oxidation of VOCs. Indeed, CeO<sub>x</sub> catalysts, possessing abundant oxygen storage and reversible Ce<sup>4+</sup>/Ce<sup>3+</sup> redox pairs, can change the oxygen release/uptake behavior during the action of oxidation catalysts, benefiting electron-involving chemical reactions.<sup>7,8</sup> However, bare CeO<sub>2</sub> materials exhibit limited catalytic activity, which can be described through two major issues: (i) low pore volume and large particle size, which restricts the access of reactants onto active centers, and/or (ii) the availability of active oxygen species that can participate in electron transfer processes during the catalytic oxidation reaction.

Heteroatom doping-driven defect engineering into the CeO<sub>2</sub> structure has emerged as a promising method to address the mentioned issues and enhance VOC catalytic oxidation. 9,10 To this end, manganese (Mn) species have been considered viable dopants that can generate numerous deficient sites. 11,12 Therefore, the formation of active oxygen species on the catalyst surface can be feasible.10 Thus, attempts have been devoted to developing cerium (Ce)-Mn binary oxide catalysts through various synthetic methods (e.g., impregnation, sol-gel, coprecipitation, redox-precipitation, urea-based combustion, and hydrothermal) directed for a wide range of applications (e.g., VOCs removal, 9,10,12-16 organic synthesis, 17 electrocatalytic ammonia synthesis,18 and CO oxidation).19 However, the prepared catalysts were unsuccessful in simultaneously addressing the two challenges posed by CeO2-based catalysts (i.e., low pore volume and active oxygen species). Combining hydrothermal and redox precipitation methods has attracted increasing attention for nanosynthesis because of the following features: (i) the formation of uniform CeOx/MnOx interfacial

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contact can induce an efficient doping effect, and (ii) rich surface adsorbed oxygen species that facilitate catalytic oxidation performance can be achieved.<sup>20,21</sup> This approach involves the participation of a reducing agent (*e.g.*, citric acid, hydrochloric acid, or organic compounds). To this end, glucose, a natural, non-toxic, and inexpensive substance, has been employed as an electron donor in the redox-precipitation method. Glucose can function as a capping agent, tailoring the growth of uniform and small-sized metal oxide nanoparticles during the synthesis.<sup>22</sup>

Herein, we attempt to incorporate Mn atoms into CeO<sub>2</sub> catalysts (Ce to Mn molar ratio of 1:1) through the glucose-assisted redox hydrothermal method followed by annealing. The characterizations reveal an interesting Mn-doped CeO<sub>2</sub> nanocatalyst (denoted as Mn-CeO<sub>2</sub>-R) that possesses a large pore volume and rich surface oxygen species, which are rarely reported elsewhere. Consequently, the as-prepared Mn-CeO<sub>2</sub>-R nanocatalyst possesses a high amount of surface oxygen species and large pore volume exhibiting outstanding performance in the total catalytic oxidation of CH<sub>3</sub>OH, HCHO, and C<sub>7</sub>H<sub>8</sub>.

# 2. Experimental

### 2.1. Materials

Paper

Cerium(III) nitrate hexahydrate (>99%, Sigma Aldrich), manganese(II) nitrate tetrahydrate (>99%, Sigma Aldrich) urea (>99%, Xilong Scientific), p-glucose anhydrous (99%, Scharlau), and KMnO $_4$  (>99%, Merck) were purchased and utilized for synthesis without further purification.

### 2.2. Material synthesis

For the glucose-assisted redox hydrothermal method, 4.35 g of Ce(NO<sub>3</sub>)<sub>3</sub>·6H<sub>2</sub>O, 1.6 g of KMnO<sub>4</sub>, and 2.4 g of glucose were added into 50 mL of double distilled water. The obtained mixture was transferred into a Teflon-line stainless steel autoclave and then sealed. The hydrothermal reaction was carried out at 150 °C for 10 hours. After the reaction, the autoclave was naturally cooled to room temperature. The obtained solid was filtered, washed with 1 L of distilled water, and then dried at 110 °C for 12 hours. The dried solid was then annealed at 400 °C under ambient conditions for 4 hours. The final Ce-Mn oxide catalyst was denoted as Mn-CeO2-R. A conventional hydrothermal method was executed with the same procedure by using urea as a precipitation agent with molar Ce-Mn-urea to 1-1-2  $(4.34 \text{ g of Ce(NO}_3)_3 \cdot 6H_2O, 2.5 \text{ g of Mn(NO}_3)_2 \cdot 4H_2O, \text{ and } 1.2 \text{ g of}$ urea are mixed in 50 mL). The solid was denoted as Mn-CeO<sub>2</sub>-C. The bare CeO<sub>2</sub> catalyst was also synthesized with the same procedure without using Mn precursors.

### 2.3. Characterization

The phase and crystal structure of bare  $CeO_2$  and Ce-Mn oxide catalysts were analyzed by a Smartlab X-ray diffractometer (XRD, Rigaku) equipped with a  $CuK\alpha$  anode. Morphologies of the catalysts were observed by JSM-6010 Plus/LV scanning electron microscope (SEM, Jeol) and a high-resolution transmission electron microscope (HRTEM, JEM 2100, Joel). The BET

(Brunauer–Emmett–Teller) surface area, micropore surface area  $(S_{\rm mp})$ , micropore volume  $(V_{\rm mp})$ , and pore distribution of the catalysts, which were previously degassed at 150 °C for 2 hours, were measured by the Asap2020 analyzer (Micromeritics). The  $S_{\rm mp}$  and  $V_{\rm mp}$  values were calculated by the t-plot method. The hydrogen temperature-programmed reduction (H<sub>2</sub>-TPR) experiments were conducted on 0.04 g of the Ce–Mn oxide catalysts by the same procedure previously published.<sup>23</sup> Thermogravimetry (TGA) analyses were performed in nitrogen (99.999%) and air atmosphere by STA6000 analyzer (PerkinElmer). Raman spectra were recorded by a Raman microscope (Xplora plus, Horiba) with an excitation laser of 532 nm.

### 2.4. Catalytic test

Catalytic total oxidation of CH<sub>3</sub>OH, HCHO, or C<sub>7</sub>H<sub>8</sub> over the Ce-Mn catalysts was conducted in a continuous-flow stainless steel BTRS microreactor (9 mm-inner diameters, Parker). 0.1 g of each catalyst was loaded into the reactor, degassed at 250 °C for 2 hours in a dried air flow, and then cooled to 30 °C. VOC vapor was generated by passing a nitrogen stream with a suitable flow rate into a thermostat bubble system containing C7H8 or formalin liquid. The VOC-containing air stream (300 ppm HCHO, 40 ppm CH<sub>3</sub>OH, or 1000 ppm C<sub>7</sub>H<sub>8</sub>) was fed into the microreactor at a flow rate of 100 mL min<sup>-1</sup> and a gas hourly space velocity (GHSV) of 60 000 mL g<sup>-1</sup> h<sup>-1</sup>. The catalyst temperature was controlled from 30-400 °C with a heating ramp of 1 °C min<sup>-1</sup>. The feed and reaction products were analyzed by an online gas chromatograph (7890B, Agilent) equipped with a flame ionization detector (FID) and a thermal conductivity detector (TCD). The conversion of  $CH_3OH$ , HCHO, and  $C_7H_8(\eta)$ was calculated by the formula (1).

$$\eta = \frac{[\text{VOCs}]_{\text{in}} - [\text{VOCs}]_{\text{out}}}{[\text{VOCs}]_{\text{in}}} \times 100$$
 (1)

where [VOCs]<sub>in</sub> and [VOCs]<sub>out</sub> are representative of the concentration of CH<sub>3</sub>OH, HCHO, or C<sub>7</sub>H<sub>8</sub> in the inlet and outlet stream, respectively.

# 3. Result and discussion

The incorporation of Mn species into the CeO<sub>x</sub> parent structure is conducted through a glucose-assisted redox hydrothermal route, followed by calcination at 400 °C for 4 hours, as shown in Scheme 1 (see Section 2.2 for details). In this circumstance, KMnO<sub>4</sub> is used as a Mn precursor and a strong oxidant, while Ce<sup>3+</sup> and glucose act as Ce precursor and reductant, respectively, to generate homogeneous dispersion of MnO<sub>2</sub> and Ce(OH)<sub>4</sub>/CeO<sub>2</sub>. CeO<sub>2</sub> nanoparticles are hydrothermally formed through the hydrolysis reactions of Ce<sup>3+</sup>/Ce<sup>4+</sup> cations. The assynthesized nano-species (MnO<sub>2</sub>, CeO<sub>2</sub>) are transformed into Mn-doped CeO<sub>2</sub> catalysts during the calcination step. The synthesis reactions could be executed as follows:

$$8MnO_4^- + C_6H_{12}O_6 \rightarrow 8MnO_2 + 6CO_2 + 8OH^- + 2H_2O$$
 (2.1)

$$3Ce^{3+} + MnO_4^- + 6H_2O \rightarrow MnO_2 + 3Ce^{4+} + 4OH^-$$
 (2.2)

Autoclave

Ce O Mn VOC

Ce(NO<sub>3</sub>)<sub>3</sub>.6H<sub>2</sub>O

KMnO<sub>4</sub>

Glucose

Mn-CeO<sub>2</sub>-R

Kich active oxygen species

Scheme 1 The illustration of the glucose-assisted redox hydrothermal synthesis of Mn-doped CeO<sub>2</sub>.

$$Ce^{4+} + 4OH^{-} \rightarrow Ce(OH)_{4}$$
 (2.3)

$$Ce(NO_3)_4 + 4H_2O \rightarrow Ce(OH)_4 + 4NO_3^- + 4H^+$$
 (2.4)

$$Ce(NO_3)_3 + 3H_2O \rightarrow Ce(OH)_3 + 3NO_3^- + 3H^+$$
 (2.5)

$$2\text{Ce}(\text{OH})_3 + 1/2\text{O}_2 \rightarrow 2\text{CeO}_2 + 3\text{H}_2\text{O}$$
 (2.6)

$$Ce(OH)_4 \rightarrow CeO_2 + 2H_2O$$
 (2.7)

In addition, the excess glucose is hydrothermally transformed to poly-carbohydrates and gluconic acid, which could act as stabilizers. The presence of glucose helps to inhibit the growth of nuclei during hydrothermal-derived recrystallization, favoring the formation of Ce–Mn nanoparticles.

### 3.1. Morphology

Scanning electron microscopy (SEM) was employed to investigate the morphology of the Mn-doped CeO<sub>r</sub> material as depicted in Fig. 1. The conventional Mn-doped CeO<sub>2</sub> (denoted as Mn-CeO<sub>2</sub>-R) and bare CeO2 samples, prepared without the addition of KMnO4 (see Section 2.2 for the synthesis details), were also prepared for comparison. Various-sized microparticles with different forms and shapes can be observed in the Mn-CeO<sub>2</sub>-C (Fig. 1A) and bare CeO<sub>2</sub> (Fig. 1B) samples. In contrast, the as-synthesized Mn-CeO<sub>2</sub>-R sample contained uniform nanoparticles (Fig. 1C and D). Fig. 1E exhibits the transmission electron microscopy (TEM) images of the prepared Mn-CeO<sub>2</sub>-R sample, confirming the existence of Mndoped CeO<sub>x</sub> nanoparticles. In addition, the HRTEM image shows the lattice spacing of 0.31 nm and 0.27 nm corresponding to the (111) and (200) planes of the CeO<sub>2</sub> phase, respectively (Fig. 1F). Such observation implies a strong impact of the preparation methods using Ce<sup>3+</sup>, glucose, and KMnO<sub>4</sub> as precursors on the formation and stabilization of the prepared catalyst with uniform nanoparticles. Decreasing material size into the nanoscale could encourage reducing the diffusion resistance of reactant molecules into active sites.

### 3.2. Phase and crystal structure

The X-ray diffraction (XRD) characterization was conducted to investigate the crystallography of the as-prepared samples as

shown in Fig. 2. The XRD pattern of the bare CeO<sub>2</sub> (line a) sample exhibits five characteristic peaks centering at  $2\theta$  of 28.6°, 33.1°, 47.4°, 56.5°, and 59.4°, which could be attributed to the (111), (200), (220), (311), and (222) crystallographic planes of the CeO<sub>2</sub> cubic fluorite structure (JCPDS card no. 00-043-1002). The Mn-CeO<sub>2</sub>-R catalyst (line b) reveals a similar XRD pattern as the bare CeO<sub>2</sub> sample without the existence of typical MnO<sub>x</sub> peaks. Furthermore, peak broadening and slight shifting could be observed. These findings suggest the insertion of Mn cations into the CeO2 lattice structure.24 The XRD pattern of Mn-CeO2-R witnesses a remarkable distortion within the CeO2 structure. Indeed, the intensity of all the characteristic diffraction peaks significantly decreases in the absence of MnO2-associated peaks. The observation could be rooted in the high degree of Ce<sup>4+</sup> substitution by Mn dopants.<sup>25</sup> Additionally, the crystallite sizes  $(D_c)$  calculated by Scherrer equations are given in Table 1. The measured  $D_c$  value of Mn-CeO<sub>2</sub>-R is found to be the smallest crystallite size (i.e., 4 nm). To this end, it can be said that the prepared Mn-CeO2-R catalyst possesses a significantly distorted structure that could offer unique surface properties (e.g., active oxygen species). In addition, its small crystal size benefits the easy access of reactant and oxygen molecules into the active sites for the catalytic oxidation of VOCs.21

Raman spectroscopy was then employed to investigate the doping effect of Mn ions within the CeO2 lattice, as depicted in Fig. 3. The spectrum of the bare CeO<sub>2</sub> sample (line a) exhibits a dominant band located at 451 cm<sup>-1</sup>, which could be attributed to the F2g symmetric stretching mode rooted in oxygen atoms surrounding Ce cations.17 Minor bands centering at  $\sim$ 240 cm<sup>-1</sup> and  $\sim$ 594 cm<sup>-1</sup> could be assigned to double acoustic (2TA) mode and defect-induced mode (D), respectively.26,27 Incorporating Mn into the ceria structure through the conventional method (i.e., the Mn-CeO2-C sample) induces a light modification within the CeO2 lattice despite a decrease in the intensity and broadening of the F2g band. The Mn-CeO2-R catalyst (line a) signifies a remarkable decrease in intensity and unambiguous red shift of the  $F_{2g}$  band toward a low Raman shift region, which could be ascribed to the replacement of Ce atoms by Mn atoms. 15,27 Furthermore, the disappearance of the D band, which is assigned to the distortion of the anionic lattice due to the formation of oxygen vacancy defects within the cubic

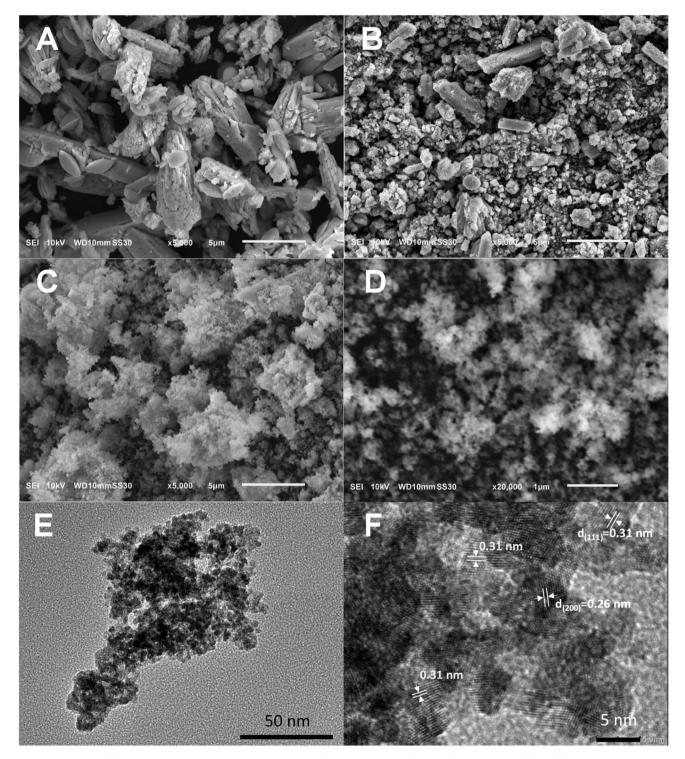


Fig. 1 SEM images of (A) bare  $CeO_2$ , and Mn-doped ceria catalysts: (B) Mn- $CeO_2$ -C, (C) and (D) Mn- $CeO_2$ -R; (E) and (F) TEM and HRTEM images of Mn- $CeO_2$ -R.

fluorite structure, could be noticed. This feature indicates that oxygen defects do not occur in Mn-CeO $_2$ -R.

### 3.3. Textural properties

Fig. 4 shows nitrogen adsorption-desorption isotherms and pore size distribution of bare  $CeO_2$  and Ce-Mn catalysts

obtained by physisorption of nitrogen at 77 K. All curves show type IV isotherm associated with  $\rm H_3$  hysteresis loop of slit-like pores-containing porous materials. <sup>28</sup> BJH pore size distribution curves (Fig. 4A–D) show peaks centering at 35 Å for bare CeO<sub>2</sub> and Mn–CeO<sub>2</sub> catalysts and 39 Å for Mn-CeO<sub>2</sub>-R catalysts, respectively. The measured BET surface area ( $S_{\rm BET}$ ), micropore

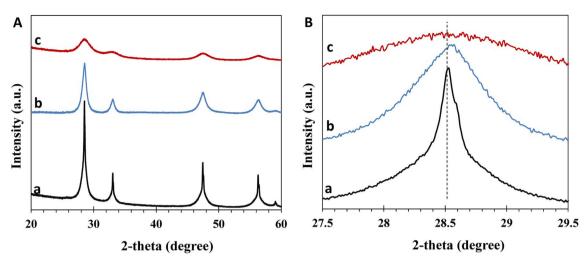


Fig. 2 XRD pattern of (a) bare CeO<sub>2</sub> and Mn-doped ceria catalysts: (b) Mn-CeO<sub>2</sub>-C and (c) Mn-CeO<sub>2</sub>-R.

Table 1 Physicochemical characteristics of the Mn-doped ceria catalysts<sup>a</sup>

Samples	$S_{\mathrm{BET}} \left( \mathrm{m}^2 \; \mathrm{g}^{-1} \right)$	$S_{\rm mp} \left( {\rm m}^2 \ {\rm g}^{-1} \right)$	$V_{\rm mp}$ (cm <sup>3</sup> g <sup>-1</sup> )	$V_{\rm p} \left({\rm cm}^3~{\rm g}^{-1}\right)$	D <sub>c</sub> (nm)	Weight $loss^b$ (%)	$\Delta_{\mathrm{WL}}$ (%)	$H_2$ consumption (mmol $g^{-1}$ )
CeO <sub>2</sub>	89	59	0.03	0.03	21.0	1.5	0.06	1.1
Mn-CeO <sub>2</sub> -C Mn-CeO <sub>2</sub> -R		50 0.7	0.02 0	0.04 0.18	10.9 4.0	2.2	0.36 0.46	1.8

<sup>&</sup>lt;sup>a</sup>  $S_{\rm mp}$ : micropore surface area,  $V_{\rm mp}$ : micropore volume,  $V_{\rm p}$ : pore volume,  $D_{\rm c}$ : crystallite size. <sup>b</sup> Weight loss corresponding to the removal of surface lattice oxygen;  $\Delta_{\rm WL}$ : difference in weight loss at 400 °C under both nitrogen and air atmosphere.

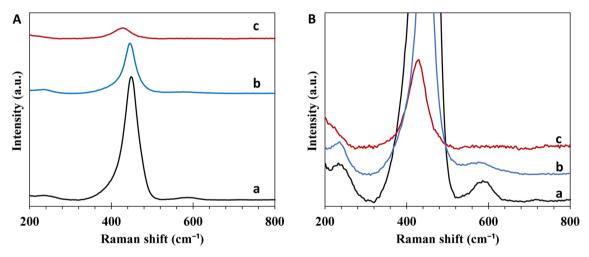


Fig. 3 Raman spectra of general (A) and extended regions (B) of (a) bare CeO<sub>2</sub>, and Mn-doped ceria catalysts, (b) Mn-CeO<sub>2</sub>-C, and (c) Mn-CeO<sub>2</sub>-R.

surface area ( $S_{\rm mp}$ ), micropore volume ( $V_{\rm mp}$ ), and pore volume ( $V_{\rm p}$ ) calculated by the desorption branch are given in Table 1. The  $S_{\rm BET}$ ,  $S_{\rm mp}$ , and  $V_{\rm p}$  values of the bare CeO<sub>2</sub> samples were found to be 89 m<sup>2</sup> g<sup>-1</sup>, 59 m<sup>2</sup> g<sup>-1</sup>, and 0.03 cm<sup>3</sup> g<sup>-1</sup>, respectively. The micropore volume of the CeO<sub>2</sub> sample is almost equal to its total pore volume, indicating that a majority of micropores having a diameter of less than 2 nm occupy its porous volume. The incorporation of Mn into the CeO<sub>2</sub> structure of Mn-CeO<sub>2</sub>-C via the conventional method slightly changes these values.

Interestingly, the Mn-CeO<sub>2</sub>-R sample exhibits a significant reduction in the  $S_{\rm BET}$  (58 m² g<sup>-1</sup>),  $S_{\rm mp}$  (0.7 m² g<sup>-1</sup>), and  $V_{\rm mp}$  (0 cm³ g<sup>-1</sup>), whereas  $V_{\rm p}$  remarkably increases to 0.18 cm³ g<sup>-1</sup>, which is 9 and 4.5 times higher than that of bare CeO<sub>2</sub> and Mn-CeO<sub>2</sub>-C, respectively. The suppression of micropores and enlargement of pore volume benefit the intraparticle diffusion of reactants and products during catalytic reactions at high temperatures. To this end, it can be said that the outstanding textural properties of the as-prepared Mn-CeO<sub>2</sub>-R, which is

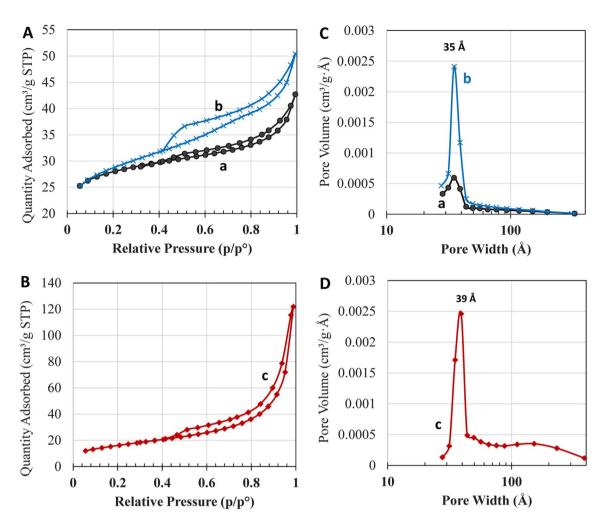


Fig. 4 (A and B) Nitrogen-isotherm adsorption curves, and (C and D) pore size distribution of (a) bare  $CeO_2$  and Mn-doped ceria catalysts, (b) Mn- $CeO_2$ -C, and (c) Mn- $CeO_2$ -R.

difficult to be obtained through conventional methods, could promote the oxidation of VOCs.

### 3.4. Thermogravimetric/differential thermal analysis

Thermogravimetric (TG) analysis in a nitrogen atmosphere was carried out to explore the evolution of surface lattice oxygen of bare CeO<sub>2</sub> and Mn-doped ceria catalysts. As shown in Fig. 5A, the TG and corresponding derivative thermogravimetry (DTG) curve of bare CeO<sub>2</sub> shows three weight loss steps: (i) at T < 160 ° C corresponding to the removal of water that had been physicochemically adsorbed on the catalyst; (ii) from 160-400 °C along with one peak at 214 °C and a shoulder at 280 °C on the corresponding DTG curve is attributed to the loss of chemically adsorbed oxygen and carbonate at the surface; (iii) from 400-780 °C with an equivalent DTG peak at 470 °C (Fig. 5A), which is assigned to the removal of surface lattice oxygen atoms, which are much weaker than that of bulk oxygen.29 At 900 °C, the phase composition of the sample is CeO2, as confirmed by the XRD pattern of the samples obtained from the TGA analysis (Fig. S1†). The thermogravimetric profiles of the Mn-CeO<sub>2</sub>-C catalyst signify a peak at 478 °C, which can be ascribed to the

removal of surface lattice oxygen (Fig. 5B). Two additional DTG peaks at 600-675 °C and 870 °C have been attributed to the decomposition of Mn(IV,III) oxides into Mn<sub>3</sub>O<sub>4</sub> at the surface and in bulk structure, respectively, which can be confirmed by the XRD measurements of the obtained sample after performing TGA analysis, as shown in Fig. S1A and B.† Regarding the Mn-CeO2-R sample, the removal of the surface lattice oxygen peak is broad and shifts to 545 °C (Fig. 5C). The amount of surface lattice oxygen available on the Ce-Mn catalyst, which is determined from the weight loss values during this second step, can be ordered as Mn-CeO<sub>2</sub>-R (2.2%) > Mn-CeO<sub>2</sub>-C (1.7%) > CeO<sub>2</sub> (1.5%) (Table 1). This indicates that Mn doping enhances the active lattice oxygen for the catalytic oxidation process. The Mn-CeO2-R could release the highest content of surface oxygen species for the total oxidation of VOCs. It has been proven that the oxygen molecules located at the surface contribute to the catalytic cycle of the oxidation process.4,12,16

The TGA analyses are also performed in the presence of air to characterize the available oxygen vacancies in the catalysts by comparing them with those obtained under a nitrogen atmosphere. Fig. 6A–C show TGA profiles of the bare CeO<sub>2</sub> and Ce–

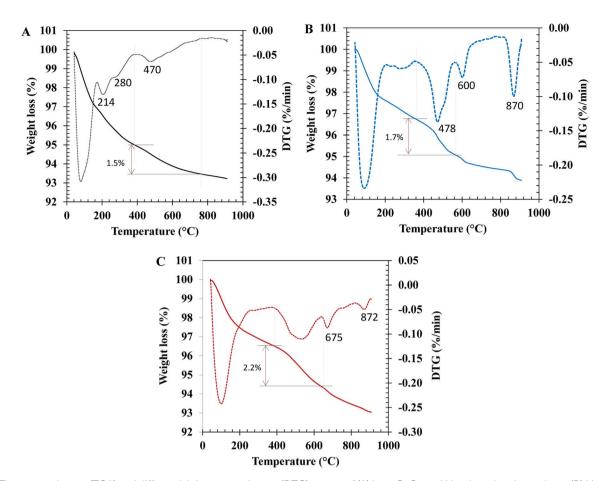


Fig. 5 Thermogravimetry (TGA) and differential thermogravimetry (DTG) curves of (A) bare  $CeO_2$  and Mn-doped ceria catalysts: (B) Mn- $CeO_2$ -C and (C) Mn- $CeO_2$ -R under nitrogen atmosphere.

Mn oxide catalysts carried out in both nitrogen and air atmosphere. At T < 800 °C, the weight loss curves evolve in nearly the same trends. The lower weight loss under an oxygen atmosphere indicates the penetration of oxygen molecules from the gas phase into the position of oxygen vacancies of ceria surface and structure to balance the weight loss due to the release of surface and bulk oxygen molecules with temperature. The weight loss difference ( $\Delta_{WL}$ ), which is calculated by the formula  $\Delta_{\mathrm{WL}} = \Delta_2 - \Delta_1$ , where  $\Delta_1$  and  $\Delta_2$  represent the difference in weight loss in an oxygen and nitrogen atmosphere at 200 °C and 440 °C, respectively, is employed as a means of the semiquantitative determination of surface oxygen vacancies available on the Ce–Mn oxide catalysts. The  $\varDelta_{\rm WL}$  values can be arranged as  $CeO_2$  (0.46%) > Mn-CeO<sub>2</sub>-C (0.36%) > Mn-CeO<sub>2</sub>-R (0.06%) (Table 1). Hence, the Mn-CeO<sub>2</sub>-R catalyst contains fewer oxygen vacancies than CeO2, which agrees with the Raman interpretation.

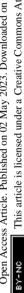
### 3.5. H<sub>2</sub>-TPR measurement

 $\rm H_2$ -TPR experiments are conducted to explore the amount of active surface oxygen in the bare  $\rm CeO_2$  and prepared Mn-doped  $\rm CeO_2$  catalysts as depicted in Fig. 7. The  $\rm H_2$ -TPR signal of bare  $\rm CeO_2$  (line a) reveals two large peaks located at 300–500 °C and

600–800 °C, which are attributed to the reduction of surface lattice oxygen species on the surface and bulk lattice oxygen in CeO<sub>2</sub>, respectively.<sup>30</sup> These peaks appear at lower temperatures in the Mn-CeO<sub>2</sub>-C (line b) sample, suggesting the higher mobility of lattice oxygen species at both surfaces and in the bulk structure.<sup>31</sup> Additionally, two peaks arising at 244 °C and 333 °C are attributed to the reduction of adsorbed oxygen species at the catalyst surface. Such oxygen species may be due to the mutual interaction between Ce and Mn in the fluorite structure.<sup>12</sup> This phenomenon could be observed at 200–350 °C in the H<sub>2</sub>-TPR signal of the Mn-CeO<sub>2</sub>-R sample. In this circumstance, a high amount of adsorbed oxygen and lattice oxygen species at the surface are crucial factors that contribute to high catalytic activity in total oxidation reactions.

The low temperature-hydrogen consumption (T < 500 °C) could convey active oxygen species as presented in Table 1. The oxygen reactivity of the prepared materials can be classified as Mn-CeO<sub>2</sub>-R (1.8 mmol g<sup>-1</sup>) > Mn-CeO<sub>2</sub>-C (1.7 mmol g<sup>-1</sup>) > CeO<sub>2</sub> (1.1 mmol g<sup>-1</sup>), respectively. Accordingly, the Mn-CeO<sub>2</sub>-R catalyst offers the largest amount of active oxygen at the catalyst surface, which is extremely reactive and energetic for reduction at relatively low temperatures during the catalytic total oxidation reaction.

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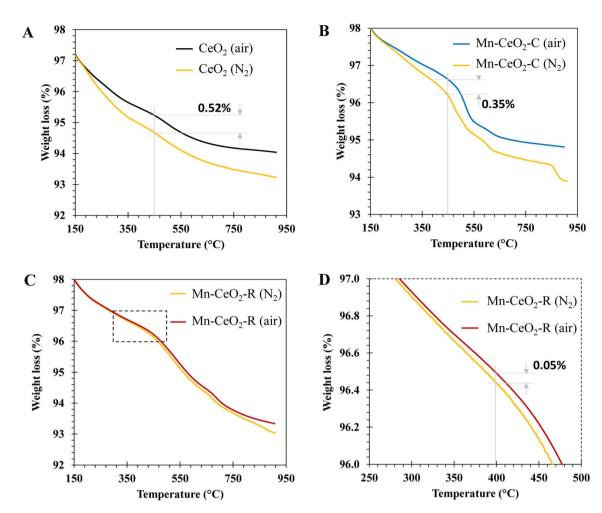


Fig. 6 Thermogravimetry (TGA) curves of (A) bare CeO<sub>2</sub> and Mn-doped ceria catalysts: (B) Mn-CeO<sub>2</sub>-C; (C) and (D) Mn-CeO<sub>2</sub>-R under nitrogen and air atmosphere

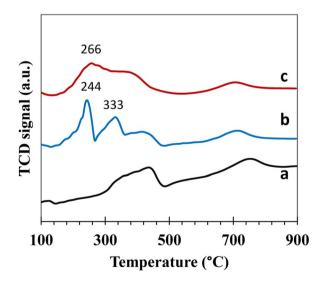


Fig. 7 H<sub>2</sub>-TPR profiles of (a) bare CeO<sub>2</sub>, and Mn-doped ceria catalysts: (a) bare CeO<sub>2</sub>, (b) Mn-CeO<sub>2</sub>-C, and (c) Mn-CeO<sub>2</sub>-R.

### Catalytic performance of total oxidation of VOCs

The utilized characterizations have pointed out three features of the prepared Mn-CeO<sub>2</sub>-R catalysts: (i) small nanoparticle size, (ii) enhanced textural properties, and (iii) rich surface oxygen species, which could significantly promote the total catalytic oxidation reactions. Motivated by these properties, two small molecules, CH<sub>3</sub>OH (kinetics diameter; d = 3.8 Å) and HCHO (d= 3.7 Å), were employed as model VOCs to evaluate the catalytic performance. Fig. 8A and B show the light-off curves representing the conversion of CH<sub>3</sub>OH and HCHO versus temperature, respectively, which reach 100% at 180-240 °C, obtained over the Ce-Mn catalysts. The conversion curves of CH<sub>3</sub>OH and HCHO over Mn-doped ceria catalysts are located at much lower temperatures than that of CeO2, indicating the strong enhancement of catalytic activities. At maximum conversion, 100% of the reactant molecules are converted into  $CO_2$ . The  $T_{50}$ and  $T_{90}$  values, *i.e.*, the temperatures at which 50% and 90% of CH<sub>3</sub>OH or HCHO are converted, are interpolated from the lightoff curves to evaluate the catalytic activity of the total oxidation of CH<sub>3</sub>OH and HCHO as summarized in Table 2. The lower  $T_{50}$ and  $T_{90}$  values indicate better catalytic activity. The catalytic performance of the prepared catalysts for HCHO oxidation is

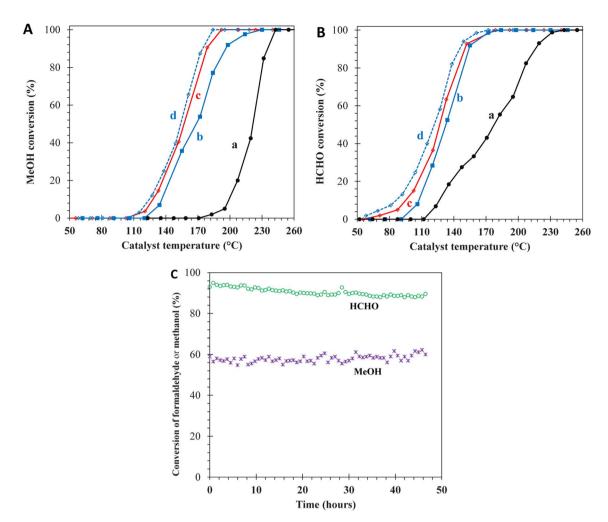


Fig. 8 Light-off curves of (A) MeOH conversion; (B) HCHO conversion over the catalysts: (a) bare  $CeO_2$ , (b) Mn- $CeO_2$ -C, (c) and (d) Mn- $CeO_2$ -R at GHSV = 60 000 mL  $g^{-1}$  h $^{-1}$ , and 30 000 mL  $g^{-1}$  h $^{-1}$ , respectively; (C) stability of HCHO and  $CH_3OH$  at 155 °C, GHSV = 60 000 mL  $g^{-1}$  h $^{-1}$ .

arranged as Mn-CeO<sub>2</sub>-R ( $T_{50} = 127$  °C;  $T_{90} = 150$  °C) > Mn-CeO<sub>2</sub>-C (133 °C; 154 °C) > CeO<sub>2</sub> (178 °C; 216 °C). A similar trend is observed in the catalytic CH<sub>3</sub>OH oxidation reaction. Therefore, it can be said that the Mn-CeO<sub>2</sub>-R sample significantly outperforms the remaining materials in the catalytic oxidation of HCHO and CH<sub>3</sub>OH compounds. Furthermore, the Mn-CeO<sub>2</sub>-R catalyst delivers reaction rates of 0.75 mmol g<sup>-1</sup> h<sup>-1</sup> and

0.04 mmol g<sup>-1</sup> h<sup>-1</sup> at GHSV of 60 000 mL g<sup>-1</sup> h<sup>-1</sup> in CH<sub>3</sub>OH and HCHO oxidation at 150 °C, respectively. The prepared Mn-CeO<sub>2</sub>-R catalyst maintains its crystalline structure during these reactions at 155 °C, corresponding to 88% and 58% conversion in the oxidation of HCHO and CH<sub>3</sub>OH for 46 hours, as depicted in Fig. S2.† At low GHSV of 30 000 mL g<sup>-1</sup> h<sup>-1</sup>, the  $T_{90}$  values reduce to 145 °C for HCHO and 175 °C for CH<sub>3</sub>OH.

Table 2 Catalytic performances of the catalysts toward oxidation of methanol, formaldehyde, and toluene

	Methanol			Formaldehyde			Toluene			
Catalysts	<i>T</i> <sub>50</sub> (°C)	<i>T</i> <sub>90</sub> (°C)	Reaction rate at 150 °C (mmol $g^{-1} h^{-1}$ )	<i>T</i> <sub>50</sub> (°C)	<i>T</i> <sub>90</sub> (°C)	Reaction rate at 150 °C (mmol $g^{-1} h^{-1}$ )	<i>T</i> <sub>50</sub> (°C)	<i>T</i> <sub>90</sub> (°C)	Reaction rate at 350 °C (mmol $g^{-1} h^{-1}$ )	$E_{\rm a}$ (kJ mol <sup>-1</sup> )
$CeO_2$	222	235	0	178	216	0.25	232	390	2.30	10.9
Mn-CeO <sub>2</sub> -C	168	196	0.03	133	154	0.72	259	360	2.36	23.7
Mn-CeO <sub>2</sub> -R	157	178	0.04	127	150	0.75	278	315	2.54	27.7
Mn-CeO <sub>2</sub> -R <sup>a</sup>	154	175	0.04	122	145	0.76				

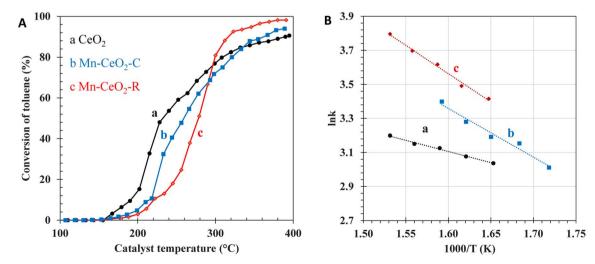
<sup>&</sup>lt;sup>a</sup> Catalytic test experiment conducted at GHSV of 30 000 mL g<sup>-1</sup> h<sup>-1</sup>.

The next attempt is to evaluate the catalytic performance toward the total oxidation of larger molecules (e.g.,  $C_7H_8$ , d=5.9 Å). Fig. 9 shows the light-off curve, which is representative of the conversion of  $C_7H_8$  *versus* the catalyst temperature. No byproduct is detected by TCD or FID detectors. The selectivity of  $CO_2$  is nearly 100%. Once again, the catalytic performance of the Mn-CeO<sub>2</sub>-R catalyst is significantly enhanced in comparison to the bare  $CeO_2$  and Mn-CeO<sub>2</sub>-C catalysts, when the temperature is above 300 °C. The obtained  $T_{90}$  values can be arranged as Mn-CeO<sub>2</sub>-R (315 °C) < Mn-CeO<sub>2</sub>-R (360 °C) < bare  $CeO_2$  (390 °C). This finding is consistent with the number of surface oxygen species, crystallite size, nanoparticles sizes, and pore volume. At high temperatures, the Mn-CeO<sub>2</sub>-R catalyst exhibits the best catalytic performance toward the total oxidation of  $C_7H_8$ , offering a reaction rate of 2.54 mmol g<sup>-1</sup> h<sup>-1</sup> at 350 °C.

At T < 300 °C, the  $T_{50}$ -derived  $C_7H_8$  conversion of the Mn-CeO<sub>2</sub>-R catalyst is lower than that of Mn-CeO<sub>2</sub>-C and bare CeO<sub>2</sub> catalysts, which is consistent with their BET surface area and amount of oxygen vacancies. Indeed, the bare CeO2 material possessing rich oxygen vacancies, as revealed by TGA and Raman, displays better oxidation of C7H8 at low C7H8 conversion and low temperatures. However, the abundance of micropores and very low pore volume of CeO2 and the Mn-CeO2-C samples restrict the conversion of C<sub>7</sub>H<sub>8</sub> at high temperatures, which could originate from the limitation of inner diffusion of reactants/products into/from the active sites. The Arrhenius plots are representative of the  $\ln k$  as a function of 1000/T, where T and k are the temperatures of the catalysts and the rate constant of the C<sub>7</sub>H<sub>8</sub> oxidation reaction, respectively. The plots are constructed at high C7H8 conversion greater than 80% and temperature above 300 °C. It is assumed that the total oxidation of C7H8 follows a first-order reaction kinetics with respect to C<sub>7</sub>H<sub>8</sub>. Moreover, the obtained energy activation has been highlighted to understand the intra and extra-particle diffusion of the Ce-Mn oxide catalysts, as shown in Fig. 8B. The obtained activation energies (E<sub>a</sub>) of CeO<sub>2</sub>, Mn-CeO<sub>2</sub>-C, and Mn-CeO<sub>2</sub>-R are 10.9 kJ mol<sup>-1</sup>, 23.7 kJ mol<sup>-1</sup>, and 27.7 kJ mol<sup>-1</sup>, respectively

(Table 2). As a result, the  $E_{\rm a}$  value observed for the  ${\rm CeO_2}$  catalyst is close to that of the diffusion-controlled step ( $\sim$ 10 kJ mol<sup>-1</sup>). This indicates that diffusion significantly impacts the mass transfer limitation over  ${\rm CeO_2}$  at high  ${\rm C_7H_8}$  conversion.  $^{32,33}$  This is consistent with the abundant micropores, large particle size, and lowest catalytic activity of the  ${\rm CeO_2}$ -R catalyst at high conversion. Accordingly, the Mn-CeO<sub>2</sub>-R catalyst containing uniform nanoparticles and large pore volume favors the access of reactants into the reactive site and the expulsion of pore diffusion products out of the inner catalyst, leading to the higher catalytic conversion of  ${\rm C_7H_8}$ .

In general, the Mn-CeO<sub>2</sub>-R catalyst with the smallest crystallite size, largest pore volume, and the highest amount of active surface oxygen, exhibits the best catalytic performance toward the total oxidation of CH<sub>3</sub>OH, HCHO, and C<sub>7</sub>H<sub>8</sub>. Mars-Van Krevelen mechanism could be employed to explain the enhanced VOC oxidation activity of the prepared Mn-CeO<sub>2</sub>-R catalyst based on structural characterizations - activity correlation. The adsorbed VOC molecules are activated by active surface oxygen species and subsequently converted to CO2 and water. This process leaves oxygen vacancies at the catalyst surface, which are later replenished by gas-phase oxygen molecules. The high amount of surface oxygen vacancies of ceria increases the reaction rate, which is valid for the total oxidation of  $C_7H_8$  at low temperatures ( $T < 300 \, ^{\circ}C$ ) in our study. However, the catalytic performance at the high conversion of C<sub>7</sub>H<sub>8</sub> is restricted by the presence of micropores. The Mn-CeO<sub>2</sub>-R catalyst, containing a trace amount of surface oxygen vacancies, shows superior catalytic performance for the total oxidation of HCHO and CH<sub>3</sub>OH. Accordingly, the abundance of active surface oxygen, created by the incorporation of Mn into ceria catalysts, controls the total oxidation of HCHO and CH<sub>3</sub>OH. Nevertheless, the effects of porosity and nanoparticles sizes are important in the total oxidation of C<sub>7</sub>H<sub>8</sub> at high conversion. The high proportion of mesopores and large pore volume of Mndoped ceria nanoparticles, created by the doping of Mn into ceria by glucose-assisted redox hydrothermal method using



 $\textbf{Fig. 9} \quad \textbf{Light-off curves of (A)} \ C_7 \\ \textbf{H}_8 \ \textbf{conversion, (B)} \ \textbf{Arrhenius plots over the various catalysts: (a) bare $CeO_2$, (b) $Mn-CeO_2-C$, and (c) $Mn-CeO_2-R$. }$ 

KMnO<sub>4</sub> and glucose as precursors, favors the intraparticle and extra-particle diffusion.

# 4. Conclusion

This work offers a novel synthetic approach to prepare Mndoped CeO2 nanoparticle catalysts through a glucose-assisted redox hydrothermal method using cerium(III) nitrate, KMnO<sub>4</sub> and glucose as precursors. The resulting catalysts, consisting of Mn doping into ceria, are characterized by uniform nanoparticles with a small crystallite size, a large mesopore volume, and a high active surface oxygen amount. These characteristics lead to superior catalytic conversion of CH<sub>3</sub>OH and HCHO as compared to bare CeO2 and conventional Mn-doped CeO2 materials. Additionally, the enhanced pore volume and micropore suppression in the Mn-doped CeO2 nanoparticles eliminate diffusion limits and improve the catalytic activity with high  $C_7H_8$  conversion (T > 300 °C). The findings demonstrate the potential of the glucose-assisted redox hydrothermal method for preparing Mn-doped ceria catalysts for low-temperature oxidation of CH<sub>3</sub>OH, HCHO, and C<sub>7</sub>H<sub>8</sub>.

# Conflicts of interest

The authors declare that they have no known competing financial interests or personal relationships that could influence the work reported in this paper.

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# References

- 1 C. He, J. Cheng, X. Zhang, M. Douthwaite, S. Pattisson and Z. Hao, *Chem. Rev.*, 2019, **119**, 4471–4568.
- 2 A. Mellouki, T. J. Wallington and J. Chen, *Chem. Rev.*, 2015, **115**, 3984–4014.
- 3 A. Krishnamurthy, B. Adebayo, T. Gelles, A. Rownaghi and F. Rezaei, *Catal. Today*, 2020, **350**, 100–119.
- 4 Y. Lyu, C. Li, X. Du, Y. Zhu, Y. Zhang and S. Li, *Environ. Sci. Pollut. Res.*, 2020, 27, 2482–2501.
- 5 H. Huang, Y. Xu, Q. Feng and D. Y. C. Leung, Catal. Sci. Technol., 2015, 5, 2649–2669.
- 6 R. Liu, H. Wu, J. Shi, X. Xu, D. Zhao, Y. H. Ng, M. Zhang, S. Liu and H. Ding, Catal. Sci. Technol., 2022, 12, 6945–6991.
- 7 A. M. D'Angelo, A. C. Y. Liu and A. L. Chaffee, *J. Phys. Chem.* C, 2016, 120, 14382–14389.
- 8 S. Scirè, S. Minicò, C. Crisafulli, C. Satriano and A. Pistone, *Appl. Catal.*, *B*, 2003, **40**, 43–49.
- 9 J. Quiroz, J.-M. Giraudon, A. Gervasini, C. Dujardin, C. Lancelot, M. Trentesaux and J.-F. Lamonier, ACS Catal., 2015, 5, 2260–2269.

- 10 S. Kurajica, K. Mužina, G. Dražić, G. Matijašić, M. Duplančić, V. Mandić, M. Župančić and I. K. Munda, *Mater. Chem. Phys.*, 2020, **244**, 122689.
- 11 Y. Zheng, J. Zhou, X. Zeng, D. Hu, F. Wang and Y. Cui, *RSC Adv.*, 2022, **12**, 25898–25905.
- 12 X. Zhang, J. Zhao, Z. Song, W. Liu, H. Zhao, M. Zhao, Y. Xing, Z. Ma and H. Du, J. Colloid Interface Sci., 2020, 562, 170–181.
- 13 D. Delimaris and T. Ioannides, *Appl. Catal., B*, 2008, **84**, 303–312.
- 14 M. T. N. Dinh, J.-M. Giraudon, A. M. Vandenbroucke, R. Morent, N. De Geyter and J.-F. Lamonier, *Appl. Catal., B*, 2015, 172–173, 65–72.
- 15 S. Guan, Q. Huang, J. Ma, W. Li, A. T. Ogunbiyi, Z. Zhou, K. Chen and Q. Zhang, *Ind. Eng. Chem. Res.*, 2020, 59, 596– 608
- 16 X. Tang, Y. Li, X. Huang, Y. Xu, H. Zhu, J. Wang and W. Shen, *Appl. Catal.*, *B*, 2006, **62**, 265–273.
- 17 R. Zhang, L. Guo, C. Chen, J. Chen, A. Chen, X. Zhao, X. Liu, Y. Xiu and Z. Hou, *Catal. Sci. Technol.*, 2015, 5, 2959–2972.
- 18 Y. Ji, W. Cheng, C. Li and X. Liu, *Inorg. Chem.*, 2022, **61**, 28–31.
- 19 K. Kim and J. W. Han, Catal. Today, 2017, 293-294, 82-88.
- 20 J. Du, Z. Qu, C. Dong, L. Song, Y. Qin and N. Huang, Appl. Surf. Sci., 2018, 433, 1025–1035.
- 21 B. de Rivas, R. López-Fonseca, C. Jiménez-González and J. I. Gutiérrez-Ortiz, *J. Catal.*, 2011, 281, 88–97.
- 22 X. Sun, C. Zheng, F. Zhang, Y. Yang, G. Wu, A. Yu and N. Guan, *J. Phys. Chem. C*, 2009, **113**, 16002–16008.
- 23 M. T. Nguyen Dinh, C. C. Nguyen, T. L. Truong Vu, V. T. Ho and Q. D. Truong, *Appl. Catal.*, *A*, 2020, **595**, 117473.
- 24 B. Murugan, A. V. Ramaswamy, D. Srinivas, C. S. Gopinath and V. Ramaswamy, *Chem. Mater.*, 2005, 17, 3983–3993.
- 25 Y. Zhang, F. Yang, R. Gao and W.-L. Dai, Appl. Surf. Sci., 2019, 471, 767–775.
- 26 Y. Lee, G. He, A. J. Akey, R. Si, M. Flytzani-Stephanopoulos and I. P. Herman, J. Am. Chem. Soc., 2011, 133, 12952–12955.
- 27 B. M. Reddy, L. Katta and G. Thrimurthulu, *Chem. Mater.*, 2010, **22**, 467–475.
- 28 S. Brunauer, L. S. Deming, W. E. Deming and E. Teller, *J. Am. Chem. Soc.*, 1940, **62**, 1723–1732.
- 29 M. V. Grabchenko, G. V. Mamontov, V. I. Zaikovskii, V. La Parola, L. F. Liotta and O. V. Vodyankina, *Appl. Catal., B*, 2020, **260**, 118148.
- 30 H. Zhu, J. Xu, Y. Yichuan, Z. Wang, Y. Gao, W. Liu and H. Yin, *J. Colloid Interface Sci.*, 2017, **508**, 1–13.
- 31 E. Gao, G. Sun, W. Zhang, M. T. Bernards, Y. He, H. Pan and Y. Shi, *Chem. Eng. J.*, 2020, **380**, 122397.
- 32 R. J. Farrauto, in *Handbook of Industrial Chemistry and Biotechnology*, Springer International Publishing, Cham, 2017, pp. 1995–2035.
- 33 V. J. Inglezakis and A. A. Zorpas, *Desalin. Water Treat.*, 2012, 39, 149–157.