


 Cite this: *RSC Adv.*, 2023, **13**, 12080

Conductive polythiophene/graphitic-carbon nitride nanocomposite for the detection of ethanol mixing in petrol

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The automobile vehicles must be operated on fuel containing no more than 10% ethanol. Use of fuel having more than 10% ethanol may cause engine malfunction, starting and running issues, and material degradation. These negative impacts could cause irreversible damage to the vehicles. Therefore, ethanol mixing in petrol should be controlled below 10% level. The current work is the first to report sensing of ethanol mixing in petrol with reference to the variation in the DC electrical conductivity of polythiophene/graphitic-carbon nitride ($\text{PTh/gC}_3\text{N}_4$) nanocomposite. The *in situ* chemical oxidative method of polymerization was used for synthesizing PTh and $\text{PTh/gC}_3\text{N}_4$ nanocomposite. Fourier transform infrared spectroscopy (FT-IR), X-rays diffraction (XRD), thermo-gravimetric analysis (TGA), transmittance electron microscopy (TEM) as well as scanning electron microscopy (SEM) analysis were used for confirmation of the structure along with morphology of the PTh and $\text{PTh/gC}_3\text{N}_4$ nanocomposite. The thermal stability of DC electrical conductivity of PTh and $\text{PTh/gC}_3\text{N}_4$ nanocomposite were tested under isothermal and cyclic ageing condition. The sensing response of PTh and $\text{PTh/gC}_3\text{N}_4$ nanocomposite as a function of DC electrical conductivity were recorded in petrol and ethanol atmosphere. The sensing response of $\text{PTh/gC}_3\text{N}_4$ nanocomposite in petrol atmosphere was 6.1 times higher than that of PTh with lower detection limit to 0.005 v/v% of ethanol prepared in *n*-hexane.

Received 18th January 2023

Accepted 13th April 2023

DOI: 10.1039/d3ra00381g

rsc.li/rsc-advances

1. Introduction

Ethanol, as a fuel produced mostly from plants like sugarcane and maize, is a desirable alternative to gasoline for reducing reliance on fossil fuels and lowering CO_2 net emissions into the environment. Furthermore, ethanol has a greater octane rating than gasoline,¹ indicating that ethanol-gasoline blends have a higher octane rating than regular gasoline. However, according to a recent analysis, ethanol-gasoline blends increase vehicle emissions of volatile organic compounds (VOCs), predominantly acetaldehydes compounds, which act as

precursors to tropospheric ozone in urban areas.² These outcomes have caused authorities to reconsider their previous support for the use of ethanol-gasoline mixes in major cities, where a high tropospheric ozone concentration is the primary environmental concern. Use of fuel containing more than 10% ethanol may cause materials degradation, starting and operating issues, and automobile engine malfunction. Automobiles can sustain irreparable damage as a result of these negative effects. Ethanol increases the danger of groundwater along with soil pollution, owing to a rise in tank corrosion, decreasing the interfacial tension between NAPL-water and lastly by preventing biodegradation and increasing the contaminant solubility. Besides its adverse effect, ethanol blending in petrol is used extensively all over of the world therefore selectively and rapidly detection to monitoring the percentage of ethanol in petrol is highly needed.^{3,4}

Many prospective materials have been studied for their behavior in chemical sensing throughout the years, as well as the sensing mechanisms that underlie the unique interactions of vapours, chemicals as well as gases with materials that are used in sensing applications. The development of chemical/gas/vapour sensors has recently focused heavily on carbon nanomaterials, metal oxide semiconductors, conductive polymers (CPs) and their nanocomposites.⁵⁻¹⁸ Due to its extraordinary functionalities through regulated charge transfer process,

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adjustable electrical conductivities, and self-resistivity changes upon exposure to various gases, conducting polymers have long been sought after as chemiresistor materials.^{16–21} Due to its extraordinary functionalities through regulated charge transfer process, adjustable electrical conductivities along with self-resistivity which alters when exposed to variety of gases, CPs have for a long time been used as chemiresistor.^{9–13} Polythiophene (PTh) being chemically, thermally and environmentally highly durable is amongst the most researched CPs. PTh and its derivatives stood out among gas sensors owing to their distinctive doping and de-doping mechanisms that produce a change in electrical conductivity when exposed to various chemicals and gases.^{22–33}

2-Dimensional nanomaterials have captured a significant deal of attention due to its several uses in the past ten years and their extremely high surface area to volume ratio. Researchers have been fascinated by MXene, graphene, g-C₃N₄, and MoS₂ among these materials because of their remarkable capacity as sensing materials, supercapacitors, and hydrogen evolution reactions.^{34–46} Taking inspiration from the aforementioned properties, here we described our efforts in developing polythiophene (PTh) and polythiophene/graphitic-carbon nitride (PTh/g-C₃N₄) nanocomposites for ethanol sensing in petrol. This is, too are awareness, the principal attempt which employs a sensor based on PTh/g-C₃N₄ nanocomposite for detecting ethanol in petrol at ambient conditions. We have also examined the sensing performance of PTh/g-C₃N₄ nanocomposite at different v/v% of ethanol in *n*-hexane.

2. Experimental

2.1. Materials

Thiophene (E. Merck), melamine (Fisher Scientific), chloroform (Fisher Scientific), anhydrous ferric chloride (CDH), methanol (E. Merck) and acetone (E. Merck) were used in their original state. Double distilled water (DDW) was used throughout the experiments.

2.2. Preparation of graphitic-carbon nitride (g-C₃N₄)

The light-yellow g-C₃N₄ powder was made by heating melamine with heating rate 15 °C min⁻¹ upto 550 °C for 3.5 h in an alumina crucible. Then the vessel containing g-C₃N₄ powder was cooled to room temperature for further processing.⁴⁷

2.3. Preparation of polythiophene (PTh) and polythiophene/graphitic-carbon nitride (PTh/g-C₃N₄) nanocomposite

PTh and PTh/g-C₃N₄ nanocomposites were prepared by the *in situ* chemical oxidative polymerization. The first step was dissolving thiophene (2 mL) in chloroform (40 mL), followed by ultrasonication for 25 minutes. Additionally, another solution of a known amount of g-C₃N₄ (20%) was prepared by adding it to chloroform (60 mL) and ultrasonically sonicated for 30 minutes. The resulting solution containing the g-C₃N₄ sheets was then poured in the thiophene solution and this mixture was then ultrasonicated for 110 minutes. The thiophene monomers got absorbed on the g-C₃N₄ sheets during ultrasonication.

Following this, a separate solution of 16.24 g (100 mmol) ferric chloride in chloroform (100 mL) was made by stirring for 20 minutes to obtain a homogeneous mixture. Now, for polymerizing thiophene monomers on the surface of g-C₃N₄ sheets, suspension of FeCl₃ was added dropwise in thiophene and g-C₃N₄ mixture followed by constant stirring on magnetic stirrer. After 24 h of stirring, the as-prepared PTh/g-C₃N₄ nanocomposite was collected by Buchner suction funnel after draining of the residual in the solution and then washed numerous times using methanol, subsequently by double distilled water and lastly by acetone. Amid washing with methanol, the colour of the material changed dark brown from deep black. The nanocomposites were lastly dried in a vacuum oven at 60 °C for 24 hours. Finally, using a mortar and pestle, these items were ground into a fine powder. An analogous procedure was used to produce the pure PTh nanoparticles.

2.4. Morphological and structural characterization

The characterization of PTh and PTh/g-C₃N₄ nanocomposites was performed by FT-IR (PerkinElmer 1725 instrument on KBr pellets), XRD (Bruker D8 diffractometer with Cu K α radiation at 1.5418 Å), TGA {PerkinElmer (Pyris Diamond instrument)}, SEM {JEOL, JSM, 6510-LV (Japan)}, and TEM {JEM 2100, JEOL (Japan)} techniques respectively.^{21–25,30–33}

DC electrical conductivities and sensing experiments were performed by four-in-line probe instrument attached with the PID controlled oven manufactured by Scientific Equipment, Roorkee, India. The thermal stability as a function of DC electrical conductivity under isothermal and cyclic ageing environments was evaluated for all the nanocomposites. The equation which was utilized for calculating the conductivity was as follows:

$$\sigma = [\ln 2(2S/W)]/[2\pi S(V/I)] \quad (1)$$

where: *V*, *S*, *W* and *I* represent the voltage (V), probe spacing (cm), the pellet-thickness (cm), current (A) and σ represents the DC electrical conductivity (S cm⁻¹) respectively.^{22,23} 250 mg of each sample was pelletised at room temperature using a hydraulic pressure instrument which operated at a pressure of 70 kN for 3 min.^{21–25,30–33}

3. Results and discussion

3.1. FT-IR

The FT-IR spectra of PTh, g-C₃N₄, and PTh/g-C₃N₄ are presented in Fig. 1. In the spectra of PTh, the wide peak at 3432 cm⁻¹ may appear as a result of O–H stretching frequency of water molecules. The peaks at 1640 cm⁻¹ and 1345 cm⁻¹ could possibly be due to the C=C, and C–C stretching mode of vibration in the polythiophene rings respectively. The peaks at 1207 cm⁻¹ and 1056 cm⁻¹ may be because of the bending vibration in PTh. The peak around 780 cm⁻¹ reveals the out of plane C–H bond bending in polythiophene. The peak at 645 cm⁻¹ is because of the bending mode of vibration of the C–S bond of the thiophene ring. The C–S–C ring deformation mode is the cause of the band at 510 cm⁻¹.^{22–25} In the g-C₃N₄ spectrum, multiple pinnacles



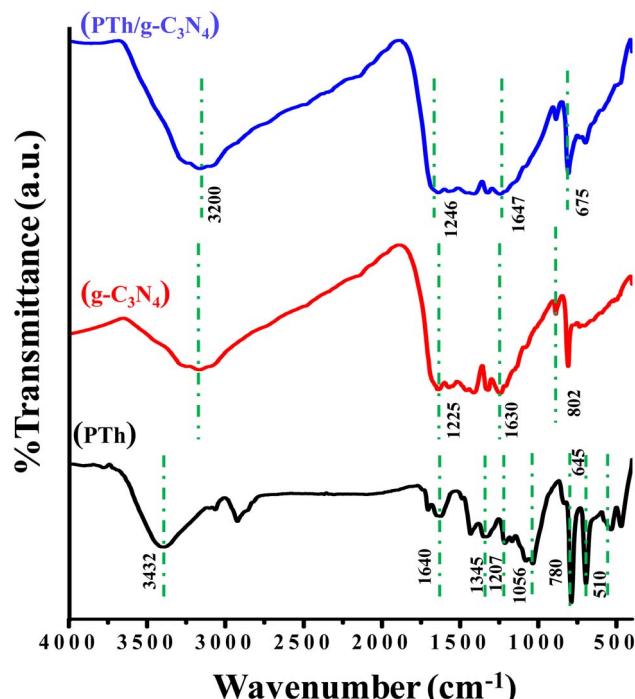


Fig. 1 FT-IR spectra of PTh, $\text{g-C}_3\text{N}_4$, and PTh/ $\text{g-C}_3\text{N}_4$ nanocomposite.

between $1225\text{--}1630\text{ cm}^{-1}$ may be due to symmetrical C–N stretching vibration and the pinnacle at 802 cm^{-1} may be traced as dissymmetrical mode of stretching of C–N and C=N as well triazine units of the aromatic rings.^{48,49} In PTh/ $\text{g-C}_3\text{N}_4$, it is found that on mixing of PTh in $\text{g-C}_3\text{N}_4$, the shoulder shifted to 1246 to 1647 cm^{-1} , affirming that the thiophene monomer gets interacted with $\text{g-C}_3\text{N}_4$. The water molecules which may be trapped by the PTh/ $\text{g-C}_3\text{N}_4$ nanocomposite and the amine groups present on $\text{g-C}_3\text{N}_4$ showing a merged peak at around 3200 cm^{-1} . The peak seen at 675 cm^{-1} may be because of the C–S bending vibrations while all other characteristics peaks of polythiophene get merged with the $\text{g-C}_3\text{N}_4$.

3.2. X-Ray diffraction studies

Fig. 2 depicts XRD patterns of $\text{g-C}_3\text{N}_4$, PTh and PTh/ $\text{g-C}_3\text{N}_4$ nanocomposite. The appeared peaks at $2\theta = 27.25^\circ$ and $2\theta = 12.87^\circ$ in $\text{g-C}_3\text{N}_4$ spectra are in accordance with the (002) and (001) Brag reflections, originated from the 2D structure of $\text{g-C}_3\text{N}_4$.^{48,49} In PTh spectra a wide hump is observed at about $2\theta = 16\text{--}27^\circ$, indicating an amorphous form of PTh.^{22\text{--}25} In case of PTh/ $\text{g-C}_3\text{N}_4$ nanocomposite a broad bump around $2\theta = 15\text{--}25^\circ$ and a sharp peak at $2\theta = 27.48^\circ$ are observed, illustrating that PTh and $\text{g-C}_3\text{N}_4$ are present in the nanocomposite. The low intensities peaks in PTh/ $\text{g-C}_3\text{N}_4$ nanocomposite clarifying that the amorphous PTh matrix might have shadowed the high intensity peaks of $\text{g-C}_3\text{N}_4$, which could be further understood by SEM and TEM studies of PTh/ $\text{g-C}_3\text{N}_4$ nanocomposite. The shift in the diffraction peak, suggested that an interaction existed between $\text{g-C}_3\text{N}_4$ and PTh in PTh/ $\text{g-C}_3\text{N}_4$ nanocomposite.

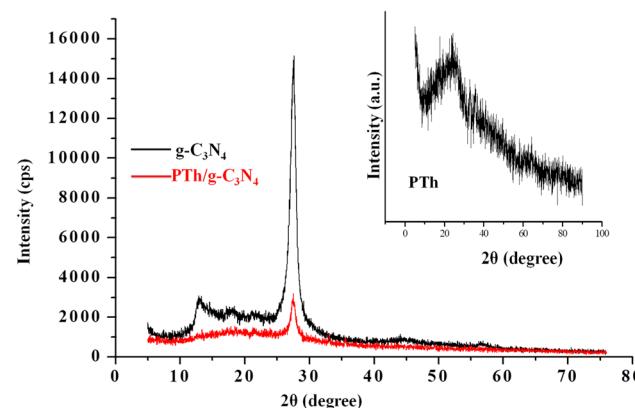


Fig. 2 XRD spectra of $\text{g-C}_3\text{N}_4$ and PTh, $\text{g-C}_3\text{N}_4$ and PTh/ $\text{g-C}_3\text{N}_4$ nanocomposite.

3.3. Scanning electron microscopy

The morphologies of PTh {Fig. 3(a)}, $\text{g-C}_3\text{N}_4$ {Fig. 3(b)} and PTh/ $\text{g-C}_3\text{N}_4$ {Fig. 3(c and d)} are depicted in Fig. 3 at different magnification by SEM operated at 15 kV. The SEM image of PTh is found to be some flaky and cloudy as shown in Fig. 3(a), while the $\text{g-C}_3\text{N}_4$ appeared in some sheet like form as shown in Fig. 3(b). The change in morphology of the PTh/ $\text{g-C}_3\text{N}_4$ nanocomposite is evident from Fig. 3(c and d), which seems entirely unlike pure PTh due to the excellent multilayer enveloping of PTh over $\text{g-C}_3\text{N}_4$ nanosheets. For the PTh/ $\text{g-C}_3\text{N}_4$ nanocomposite, PTh is well mounted on $\text{g-C}_3\text{N}_4$ nanosheets, resulting in the morphology becoming flaky and sheet-like. Absence of free $\text{g-C}_3\text{N}_4$ nanosheets are visible in the PTh/ $\text{g-C}_3\text{N}_4$ nanocomposite, it is possible to conclude that PTh effectively covers $\text{g-C}_3\text{N}_4$ nanosheets.

3.4. Transmission electron microscopy

Fig. 4 displays the TEM micrograph of PTh/ $\text{g-C}_3\text{N}_4$. Here in this image, it can be clearly observed that there is sheet of $\text{g-C}_3\text{N}_4$ upon which PTh has been successfully deposited and masked upon the entire surface of the $\text{g-C}_3\text{N}_4$ sheet. The monomers of thiophene polymerize over the large surface of $\text{g-C}_3\text{N}_4$ providing an effective π -conjugated system which promotes excessive transit of the charge carriers, which the reason for PTh/ $\text{g-C}_3\text{N}_4$ showing higher sensing performance towards ethanol.

3.5. Thermogravimetric analysis

Thermo-gravimetric analysis of PTh, $\text{g-C}_3\text{N}_4$ and PTh/ $\text{g-C}_3\text{N}_4$ nanocomposite is depicted in Fig. 5. In the degradation procedure of $\text{g-C}_3\text{N}_4$, the initial weight loss was found from 50 to 120°C due to loss of water molecules. Subsequently, the oxidative thermal degradation of $\text{g-C}_3\text{N}_4$ is started at around 510°C . While the PTh found to be least thermally stable among all of the samples. The initial loss of weight in PTh may be as a result of vaporization of water molecules followed by further degradation started around at 275°C . In case of PTh/ $\text{g-C}_3\text{N}_4$ nanocomposite, after losing of water molecules at around 80 to 105°C , the final degradation started from 350°C which is higher



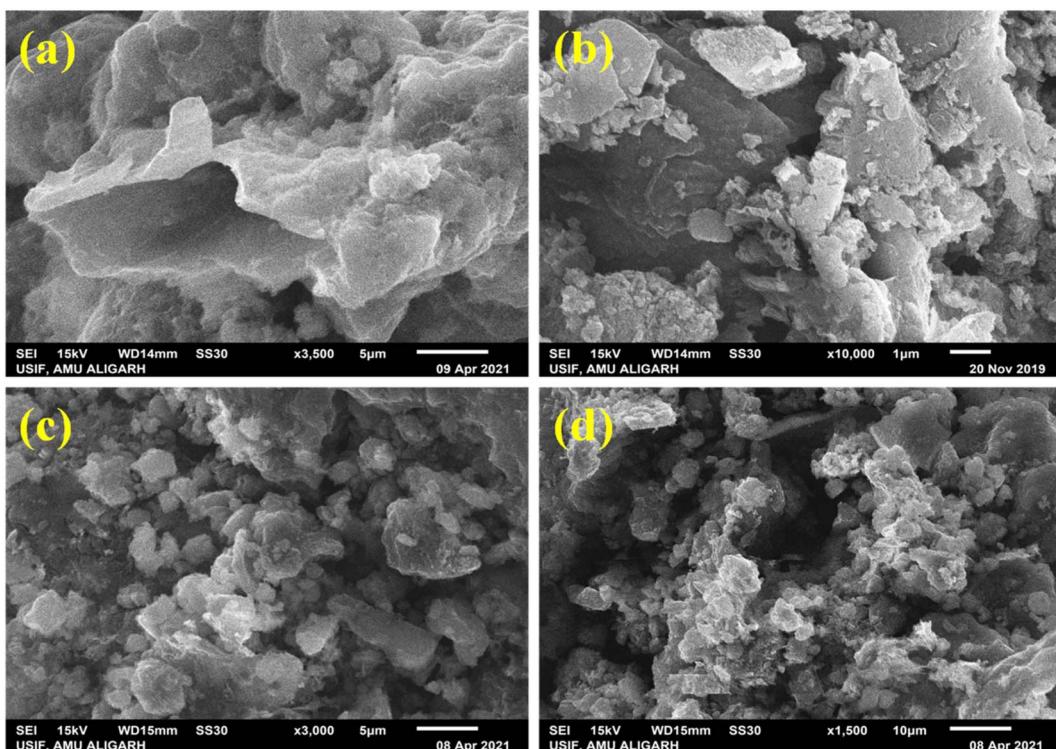


Fig. 3 SEM images of: (a) PTh, (b) g-C₃N₄, and ((c) and (d)) PTh/g-C₃N₄ nanocomposite.

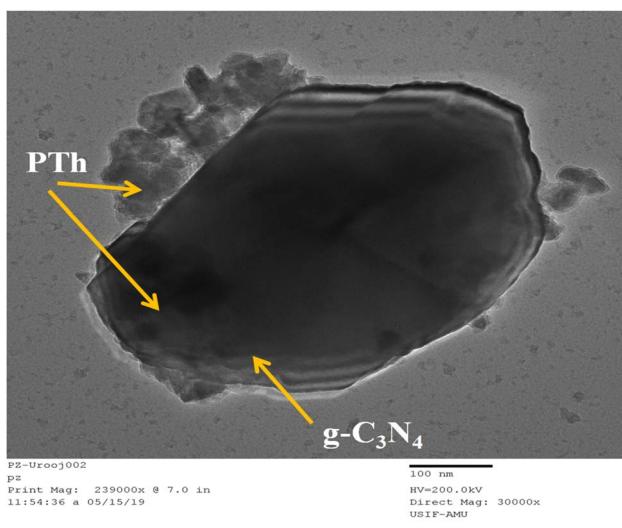


Fig. 4 TEM image of PTh/g-C₃N₄ nanocomposite.

than that of PTh but lower than g-C₃N₄, suggesting the higher thermal stability of PTh/g-C₃N₄ nanocomposite than that of PTh, which may be a result of the electronic interaction taking place amid PTh and g-C₃N₄ nanosheets.

4. Electrical conductivity studies

Fig. 6 the DC electrical conductivities of PTh, g-C₃N₄ and PTh/g-C₃N₄ nanocomposite were calculated by a 4-in-line probe

method and marked as 8.326×10^{-4} S cm⁻¹, 5.8273×10^{-5} S cm⁻¹ and 6.4327×10^{-4} S cm⁻¹ respectively as shown in Fig. 6(a). It was anticipated that upon interacting of lone pairs of S-atoms and π -electrons of thiophene ring in polythiophene with g-C₃N₄ sheets, the electrical conductivity will increase by generating more polarons on PTh as depicted in Fig. 6(b). Though, it is expected that there was a decline in electrical conductivity upon g-C₃N₄ in PTh may be due the insulating nature of g-C₃N₄. However, the fall in DC electrical conductivity was not as much and it is still found in the range of PTh, might be due to the more dominating effect of generated polarons in

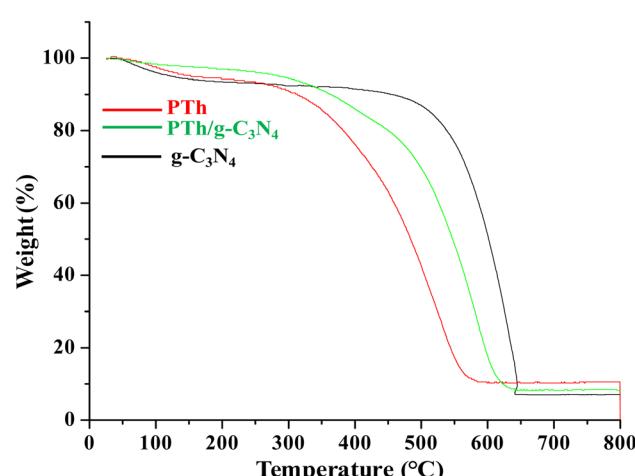


Fig. 5 Thermogravimetric analysis of PTh, g-C₃N₄ and PTh/g-C₃N₄ nanocomposite.



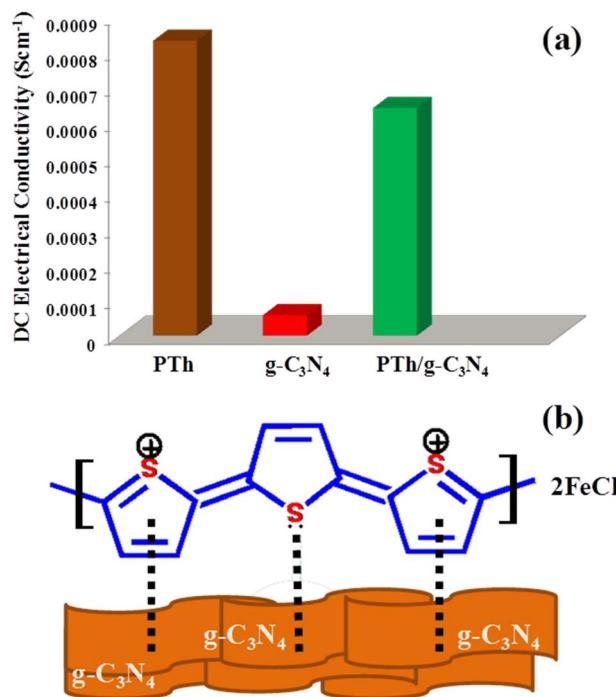


Fig. 6 (a) Initial DC electrical conductivity of PTh, g-C₃N₄ and PTh/g-C₃N₄ nanocomposite and (b) the possible interaction between PTh and g-C₃N₄ sheets in PTh/g-C₃N₄ nanocomposite.

PTh than insulating g-C₃N₄ in tug of war between them. After all, despite an insignificant loss of electrical conductivity in PTh/g-C₃N₄ nanocomposite on incorporation of g-C₃N₄ in PTh, it provides a high surface area in nanocomposite by the formation of effective arrangement of the π -conjugated system of PTh on the large g-C₃N₄ surface area.

4.1. Retention of DC electrical conductivity under isothermal ageing condition

The stability as a function of DC electrical conductivity retaining capacity under isothermal ageing conditions of the

nanocomposites PTh and PTh/g-C₃N₄ were examined, as shown in Fig. 7.

The equation applied for calculating the relative DC electrical conductivity ($\sigma_{r,t}$) at constant temperature was:

$$\sigma_{r,t} = \sigma_t / \sigma_0 \quad (2)$$

in this equation, σ_t and σ_0 is represent the DC electrical conductivities at time 't' and '0', respectively.⁵⁰⁻⁵³

It is clearly noticeable from Fig. 7(a) that PTh exhibited thermally stable electrical conductivity at temperatures 50 °C, 70 °C and 90 °C. At constant temperature up to 90 °C, DC electrical conductivity rises with time indicating the semiconducting nature of PTh. Whereas above 90 °C, there is a continuous fall of the DC electrical conductivity with time, which may be due to the damage of conducting channels as well as the loss of the doping agent in PTh. While for the PTh/g-C₃N₄, the DC electrical conductivity was found to be stable at 50 °C, 70 °C, 90 °C and 110 °C and behaves like semiconductor as displayed in Fig. 7(b). The incorporation of g-C₃N₄ in PTh stabilises its DC electrical conductivity even in high temperature condition (up to 110 °C) strongly suggesting presence of an electronic interaction between PTh and g-C₃N₄, which resists the damaging of its conducting channels and loss of the doping agent. These outcomes displayed that PTh/g-C₃N₄ could be an auspicious candidate in various electronic applications even at 110 °C.

4.2. Retention of DC electrical conductivity under cyclic ageing condition

PTh and PTh/g-C₃N₄ nanocomposite were also examined for their stability as a function of DC electrical conductivity performance under cyclic ageing conditions, as shown in Fig. 8.

The equation applied for calculating the relative DC electrical conductivity (σ_r) was:

$$\sigma_r = \sigma_T / \sigma_{50} \quad (3)$$

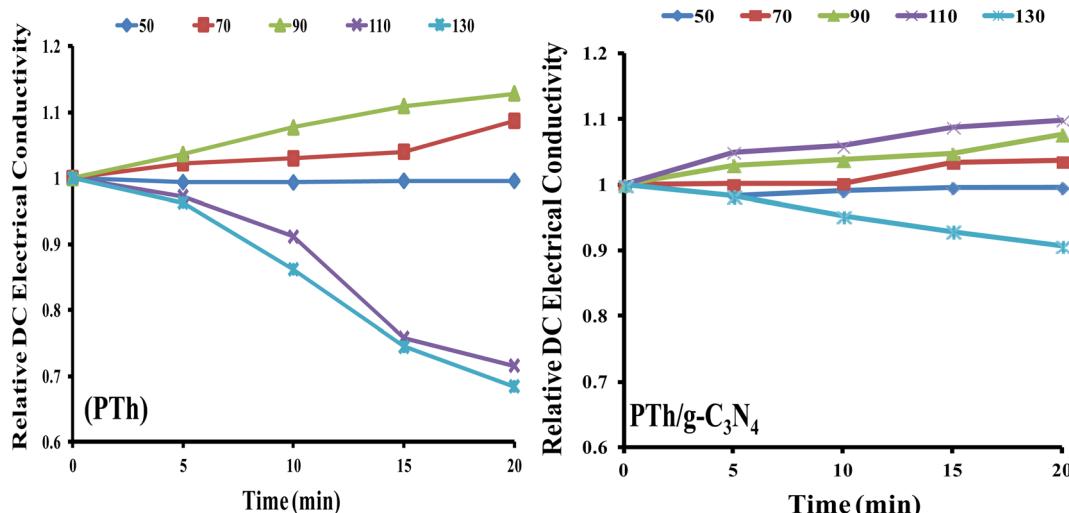


Fig. 7 Relative DC electrical conductivity of PTh, and PTh/g-C₃N₄ nanocomposite under isothermal ageing environment.



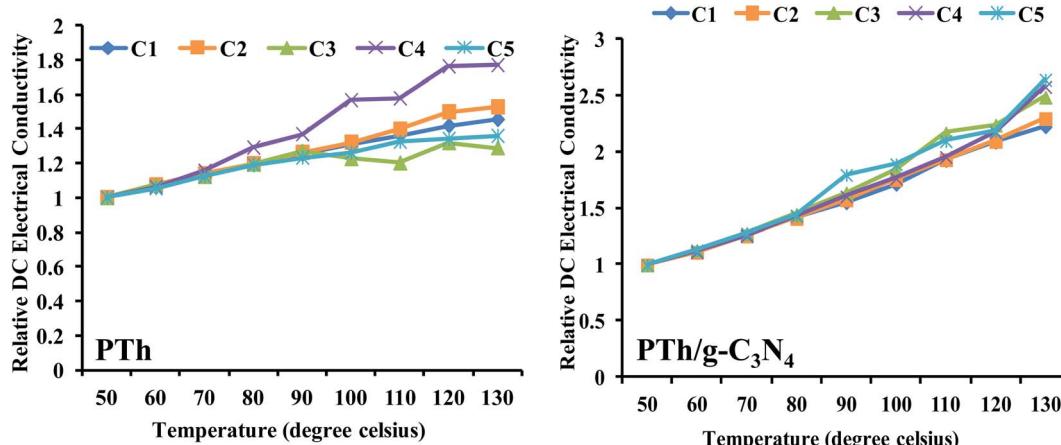


Fig. 8 Relative DC electrical conductivity of PTh, and PTh/g-C₃N₄ nanocomposite under cyclic ageing environment.

where: σ_T and σ_{50} is representative of the DC electrical conductivities at temperatures T and 50 °C.⁵⁰⁻⁵³

The DC electrical conductivity of both the samples were recorded by four consecutive cycles with continuously rise in temperature up to 130 °C. These findings revealed that DC electrical conductivity increased steadily in each cycle and followed a same pattern in both of the samples (PTh and PTh/g-C₃N₄), which may be attributed to the high movability of polarons at elevated temperatures. For PTh {Fig. 8(a)}, in the third and fourth cycles, the conductivity shows different pattern might be as a result of damage of the material and loss of proper conductive channel by repeatedly heating and cooling of PTh. While PTh/g-C₃N₄ displayed continuous rise in conductivity with each cycle without losing its conductivity, signifying the presence of electronic interaction between PTh and g-C₃N₄.

5. Sensing

The electrical conductivity of CPs depends upon many factors *e.g.*, dopants, temperature and fillers *etc.*^{17-22,50,51} In p-type doped conducting polymers, holes like polarons and bipolarons generated act as carriers of charge in the backbone of extended π -conjugated system of these CPs. The mobility of these polarons and bipolarons charge carriers could considerably be hindered by any electronic interaction with the polymer chain. Adsorption and desorption of any analyte gas on the polymer sensor surface is the basic step in gas sensing procedure to monitor the effective electrical conductivity change of the polymer.²⁰⁻²⁵ Presence of analyte gases can be detected by simple change in electrical conductivity upon adsorption on the sensor's surface, as they interact with polarons of PTh, leading to decline in the DC electrical conductivity. Therefore, here PTh and PTh/g-C₃N₄ nanocomposites were tested for change in DC electrical conductivity on simple adsorption and desorption process of the analyte gas.

The % sensing response (S) is calculated by the following equation:

$$S = (\Delta\sigma/\sigma_i) \times 100 \quad (4)$$

where, σ_i and $\Delta\sigma$ are representative of the initial DC electrical conductivity as well as change in DC electrical conductivity during complete exposure of gas respectively.^{33,50,51}

5.1. Sensing response

Sensing of ethanol in petrol was studied by examining the performance of relative DC electrical conductivity of PTh and PTh/g-C₃N₄ nanocomposite upon exposing to petrol vapours followed by ambient air as shown in Fig. 9. The PTh and PTh/g-C₃N₄ nanocomposite were finely grinded in powder form and then transformed into pellet form by hydraulic pressure machine. The pellets of each of the sample was attached to the four-in-line probes and placed in a sealed sensing chamber. At first PTh was exposed to petrol vapours followed by room atmosphere for a fixed time. In petrol vapour atmosphere, the DC electrical conductivity of PTh decreased exponentially for about first 90 s and then gets saturated for further change. The loss in electrical conductivity may be due to the ethanol present in petrol because as we know petrol comprises of mainly non-

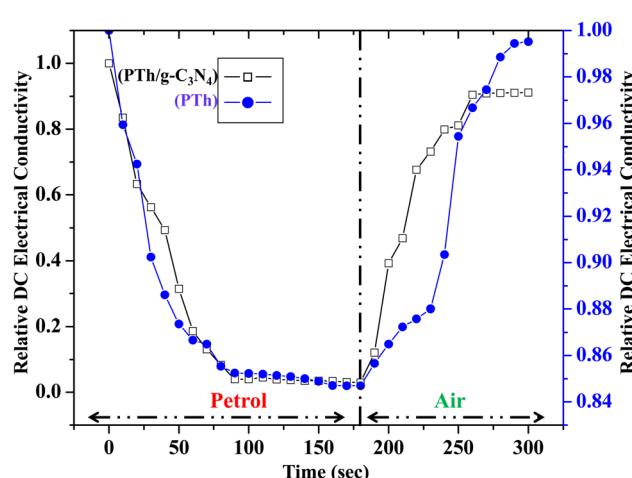


Fig. 9 Relative DC electrical conductivity of PTh and PTh/g-C₃N₄ nanocomposite in petrol atmosphere.

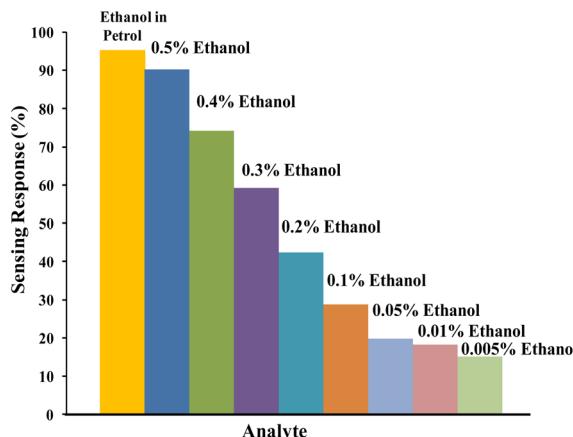


Fig. 10 Sensing response of PTh/g-C₃N₄ nanocomposite towards different v/v% of ethanol.

polar hydrocarbons with some mixing of ethanol to increase its octane rating.³³ The hydrocarbon molecules in petrol being non-polar at ambient temperatures, don't interact electronically with polarons of PTh. The ethanol (C₂H₅OH) in petrol interacts with the polarons of PTh, disrupting the conductivity channels by impeding the movement of some polarons of PTh, causing decline in DC electrical conductivity. As PTh pellet exposed to ambient air, the relative DC electrical conductivity began to increase and came back to its initial value within 95 s and then flattened may be as a reason of complete desorption of ethanol from the PTh surface. Similarly for the PTh/g-C₃N₄ nanocomposite, the relative DC electrical conductivity declined by exposing the pellet of PTh/g-C₃N₄ nanocomposite to petrol and reverted back to its initial value within 80 s in air. The sensing response of PTh and PTh/g-C₃N₄ nanocomposite were found to

be 15.55% and 94.73% respectively. The sensing response calculated in PTh/g-C₃N₄ nanocomposite was 6.1 times higher than that of PTh may be due to the high adsorption–desorption on PTh/g-C₃N₄ nanocomposite surface. Thus, incorporation of g-C₃N₄ in PTh, creating high surface area with proper advancing in conducting channels of PTh for adsorption and desorption of analyte gases. The sensing response of PTh/g-C₃N₄ toward ethanol in petrol as well as at different volume percentage (0.5%, 0.4%, 0.3%, 0.2%, 0.1% and 0.05%) of ethanol in *n*-hexane were also determined that is 95.3345%, 90.152%, 74.242%, 59.091%, 42.424%, 28.788%, 19.697%, 18.188% and 15.152% respectively as shown in Fig. 10. The sensing response towards petrol was the highest among them may due to the highest ethanol mixing in petrol. As the volume percentage of ethanol decreases, the sensing response decreases due to fewer interactions of lone pair of ethanol with polarons of PTh. As a result, one can say that considerable variation in conductivity was seen at high concentrations.

5.2. Ethanol sensing

In order to justify the PTh/g-C₃N₄ nanocomposite's sensing response towards ethanol in petrol, the DC electrical conductivity performance of PTh/g-C₃N₄ nanocomposite was also recorded for different volume percentage of ethanol in *n*-hexane as shown in Fig. 11. The DC electrical conductivity follows the same pattern as it is for petrol. As the volume percentage of ethanol increases, the variation in DC electrical conductivity of PTh/g-C₃N₄ nanocomposite increases, may be due to high interaction of ethanol with the polarons of PTh/g-C₃N₄ nanocomposite and gets saturated after few seconds. While in ambient air, conductivity completely reverted to its initial value by simple desorption of ethanol from PTh/g-C₃N₄ nanocomposite surface.

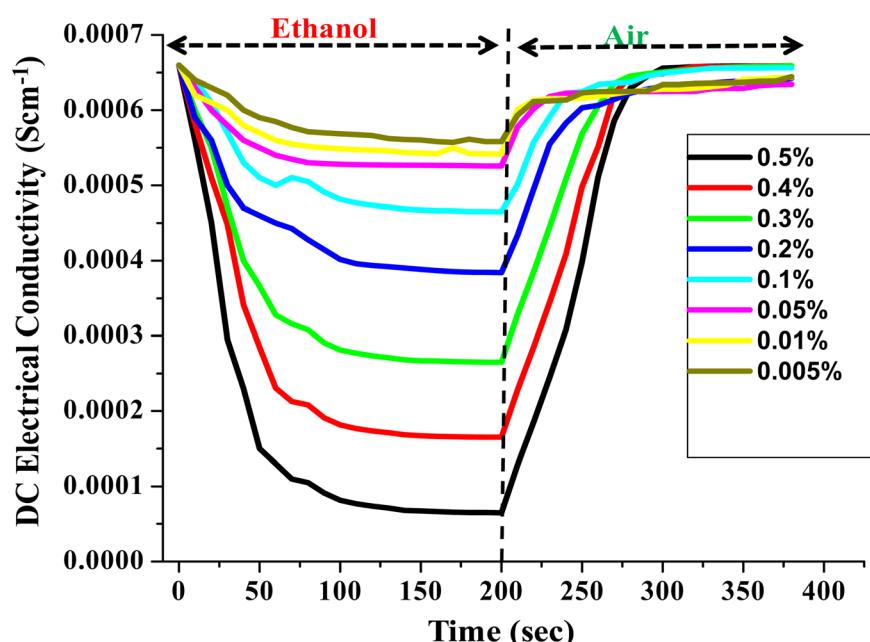


Fig. 11 Steady-state response of DC electrical conductivity of PTh/g-C₃N₄ nanocomposite in different v/v% of ethanol atmosphere.



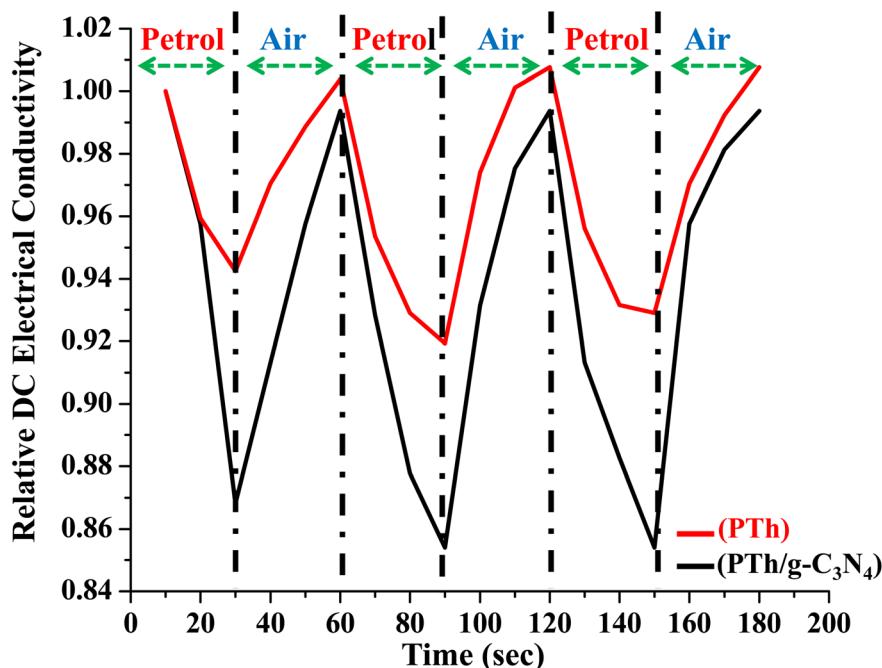


Fig. 12 Relative DC electrical conductivity of PTh and PTh/g-C₃N₄ nanocomposite on alternate exposure to petrol and ambient air with respect to time.

5.3. Reversibility test

The PTh and PTh/g-C₃N₄ nanocomposite were tested for their rapid adsorption and desorption of analyte by observing the variation in relative DC electrical conductivity in environment of petrol for 30 s followed by ambient air for the next 30 s consecutively for three 60 s cycles as shown in Fig. 12. Both of the samples showed excellent reversibility with minimum loss in DC electrical conductivity in repeated three consecutive

cycles. In PTh/g-C₃N₄ nanocomposite the variation in relative DC electrical conductivity was more superior to that of PTh, which may be due to more adsorption–desorption of analyte on PTh/g-C₃N₄ nanocomposite surface than that of PTh.

5.4. Selectivity

High selectivity is an essential criterion for a gas sensor in order to qualify as reliable and applicable. The PTh/g-C₃N₄

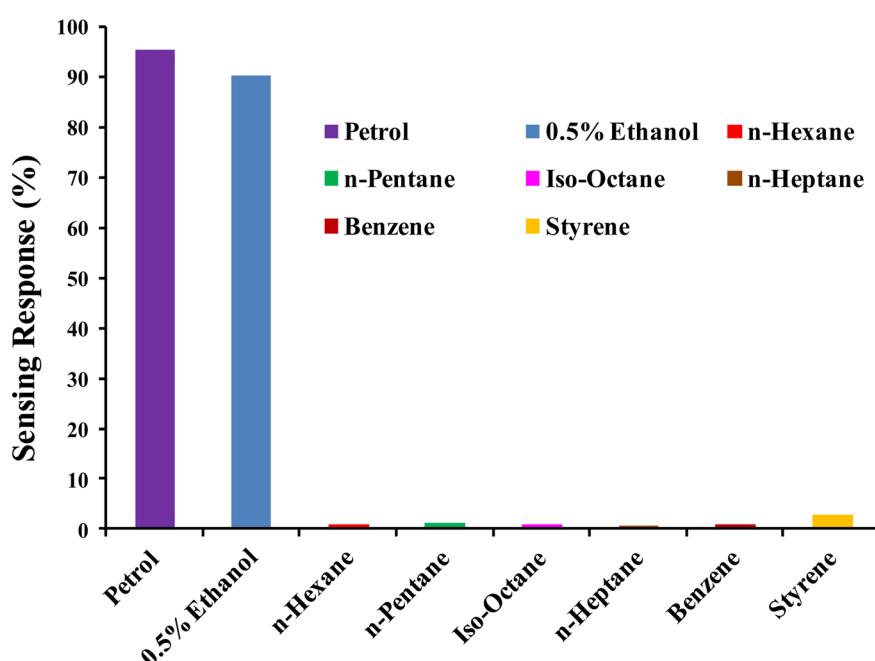


Fig. 13 Selectivity of PTh/g-C₃N₄ nanocomposite towards ethanol vs. different hydrocarbons.

nanocomposite sensing response toward ethanol mixing in petrol, 0.5% ethanol in *n*-hexane, *n*-hexane, *n*-pentane, iso-octane, *n*-heptane, benzene, styrene at room temperature (27 °C) are displayed in Fig. 13. High selective response of PTh/g-C₃N₄ nanocomposite toward ethanol may be due its availability of lone pairs of oxygen which interacts with polarons of PTh, causing significant loss of DC electrical conductivity (high sensing response). While for the other analytes (hydrocarbons) tested, the sensing responses were negligible in comparison of ethanol may be of their non-polar nature. Thus the lower the polarity/free lone pairs of electrons in the analyte, lower the conductivity change occurred in PTh/g-C₃N₄ nanocomposite, makes ethanol highly selective in comparison of different hydrocarbon compounds.

5.5. Sensing mechanism

Sensing mechanism of PTh/g-C₃N₄ nanocomposite is explained by the DC electrical conductivity behaviour on simple adsorption and desorption mechanism of analyte at room temperature, as depicted in Scheme 1. In PTh/g-C₃N₄ nanocomposite, graphitic-carbon nitride get interacted with pi-bonds of thiophene rings in polythiophene and form a nanocomposite providing high surfaced platform for sensing of analytes. It is already known that petrol is comprises of majority of hydrocarbons and some ethanol mixing to raise its octane number. The hydrocarbon molecules in petrol being nonpolar at ambient temperature, do not interact with the polarons of polythiophene in PTh/g-C₃N₄ nanocomposite. Ethanol is predicted as the only candidate that could be accountable for the variation in the conductivity of PTh/g-C₃N₄ nanocomposite upon exposure to petrol vapors. In presence of petrol atmosphere, the lone pairs of ethanol interact with the polarons of

PTh/g-C₃N₄ nanocomposite, which in turn hindered the movement of polarons, eventually declination of DC electrical conductivity with time. While in ambient air condition, the ethanol molecules and petrol gets desorbed from the sensor surface, resulting in reverting its conductivity to its about initial values, indicating complete desorption of ethanol molecules mixed in petrol. Thus, the above mechanism indicates that polarons mobility is controlled by the simple adsorption and desorption of ethanol on PTh/g-C₃N₄ nanocomposite surface.

6. Conclusions

In this paper, we synthesized PTh and PTh/g-C₃N₄ nanocomposite, characterized and studied for their stability of DC electrical conductivity and sensing response toward ethanol mixing in petrol. The stable nature of DC electrical conductivity of PTh/g-C₃N₄ nanocomposite under accelerated isothermal and cyclic ageing condition were demonstrated and found significantly higher than that of PTh. Also the PTh/g-C₃N₄ nanocomposite showed rapid response and high reversibility towards ethanol mixing in petrol and ethanol prepared in *n*-hexane with a detection limit of 0.005%. The sensing response in PTh/g-C₃N₄ nanocomposite towards ethanol mixing in petrol was 6.1 times higher than that of PTh at room temperature. In addition of excellent sensing response, this research found that PTh/g-C₃N₄ nanocomposite might be used as a semiconducting material in a variety of electronic and electrical devices at high temperature conditions.

Conflicts of interest

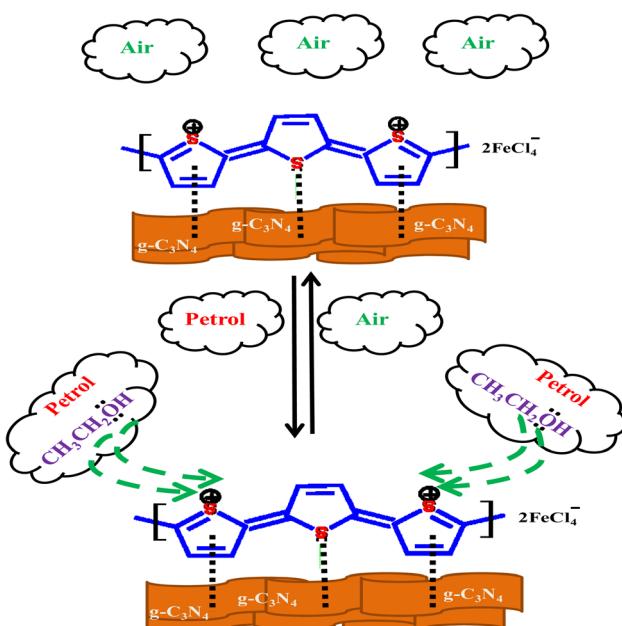
The authors declare that they have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper.

Acknowledgements

Authors are thankful to USIF, AMU, Aligarh-202002 for providing SEM and TEM facilities.

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Scheme 1 Proposed mechanism of interaction of ethanol with the PTh/g-C₃N₄ nanocomposite.



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