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The impact of oxygen on Ga doped ZnO film

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The Ga doped ZnO (GZO) film is one of the promising alternative films to replace ITO film, but its properties suffer from degradation when it is deposited under oxygen-rich conditions. This degradation has been investigated by depositing the films under different oxygen partial pressures. XRD results showed that all GZO films had wurtzite structure and the lattice parameter-*c* contracted when oxygen was introduced into the argon deposition atmosphere, but the parameter-*c* nearly remained constant when oxygen partial pressures were further increased. The contraction of parameter-*c* was caused by the increasing concentrations of V_{Zn} (Zn vacancy). It was the first time to observe that the impurity phase of $Ga_2Zn_6O_9$ appeared and disappeared in GZO films during the increase of oxygen partial pressures. Analogously, conductivity decayed and optical bandgap decreased abruptly as oxygen was introduced, which enhanced self-compensation of donors and acceptors. The energy band structures of GZO and ZnO films were determined by using UPS, and the results showed that oxygen had little effect on the electron affinity of the GZO film, but a significant difference in electron affinity between the ZnO and GZO films was observed. This result indicated that although the electron affinity of ZnO could be effectively tuned by doping with Ga, it remained quite stable for GZO under oxygen-rich conditions.

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1. Introduction

ZnO has been used as an electron transport or buffer layer in photovoltaics, and different modification methods have been applied to ZnO to achieve suitable performance. One of the most effective methods to improve conductivity is by doping ZnO with Ga. GZO is a direct wide band gap semiconductor material, and the high electron concentration is mainly due to the defect of Ga substitutions for Zn (Ga_{Zn}). Ga_{Zn} is donor defect because the defect energy level is only 0.1 eV below the conductive band minimum of ZnO, and the formation energy of Ga_{Zn} was low.¹ Besides, due to the similar ionic radii of Ga^{3+} (0.62 Å) and Zn^{2+} (0.72 Å), and comparable bond lengths with oxygen ions of Zn–O (1.97 Å) and Ga–O (1.92 Å), there would be small lattice mismatch and disorder of local structures in GZO.^{2–4} The value of resistivity for ZnO film doped with Ga with 4 at% was in the magnitude of 10^{-4} Ω cm, achieved by using RF sputtering method, which approached that of ITO film.⁵

However, GZO film has been proved to be heavily impacted by the oxygen during the growth of films. M. H. Lee *et al.* investigated the intrinsic defects of GZO by adopting *ab initio* methods, and the results showed that the formation energy (E_f) of Ga_{Zn} was −3.7 eV in an oxygen-rich condition, while it increased to −0.21 eV in an oxygen-poor condition. The decline

of intrinsic defect formation energies was $E_f(V_{Zn}) < E_f(O_i) < E_f(V_O)$ in oxygen-rich conditions, but it was $E_f(V_O) < E_f(V_{Zn}) < E_f(O_i)$ in oxygen-poor conditions.⁶ C. S. Lee *et al.* prepared $Zn_{0.96}Ga_{0.04}O$ films on the quartz substrates, and the film growth was carried out in a mixed gas of Ar and O_2 , with the $O_2/Ar + O_2$ ratios ranging from 0.0 to 1.0. The experimental results showed that with the addition of the oxygen in the growth ambient, the resistivity abruptly increased from 2.52×10^{-4} Ω cm to 0.295 Ω cm, which was derived from the sudden reductions in the electron density and mobility because of the compensation *via* $Ga_{Zn}-V_{Zn}$ complexes.⁵ The self-compensation effect between V_{Zn} defects energy level and Ga^{3+} doped energy level has been also revealed in ZnO matrix by D. C. Look.⁷

The GZO thin film transistors were fabricated by X. Q. Li *et al.*, and the GZO (2 wt% Ga) film was deposited at variable oxygen pressure. The experimental results showed that a small amount of oxygen caused a decrease in oxygen deficiency of the film, which reduced electron trapping and facilitated carrier transport. However, excessive oxygen caused an increase in the receptor complex such as $Ga_{Zn}-O_{in}$ and $Ga_{Zn}-V_{Zn}$.⁸ W. S. Noha *et al.* prepared the $ZnO:Ga_{0.01}P_{0.04}$ thin films by the PLD method, and the film exhibited a transition from n-type to p-type conductivity above the oxygen partial pressure of approximately 200 mTorr.⁹

In this work, the GZO films were prepared by using RF sputtering under various oxygen partial pressures, and the band structure, impurity and opto-electrical properties were investigated.

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2. Experimental method

GZO films were deposited on low-alkali barium-borosilicate glass (Corning7059) substrates by RF magnetron sputtering. A 3-inch diameter target of Ga-doped ZnO (with 3 at% Ga) was used, and the distance of target to the substrate was 12 cm. The sputtering power was maintained at 150 W, and the deposited pressure was 2 Pa at room temperature, with variation in oxygen partial pressures as shown in Table 1. As the reference, a ZnO film was deposited by using a Zn target in Ar + O₂ (4 : 1) sputtering atmosphere.

The structures of thin films were characterized by using X-ray diffraction (XRD), and the chemical states of the film surface were investigated by using X-ray photoelectron spectroscopy (XPS) after sputter etching of the film surface. The morphology was carried out by using scanning electronic microscopy (SEM). The electrical properties were analyzed by using the Hall effect measurement, and the transmittance of the thin films in the wavelength range of 300–900 nm was measured by using an ultraviolet-visible spectrophotometer.

3. Results and discussions

The structures of GZO films have been studied by using XRD. From Fig. 1(a), it could be observed that the G1 film exhibited a strong peak and a weak peak at 34.47° and 72.67°,

Table 1 The GZO films deposited under the different oxygen partial pressures

Sample name	G1	G2	G3	G4	G5	G6
Oxygen partial pressure/Pa	0	0.065	0.125	0.182	0.235	0.286

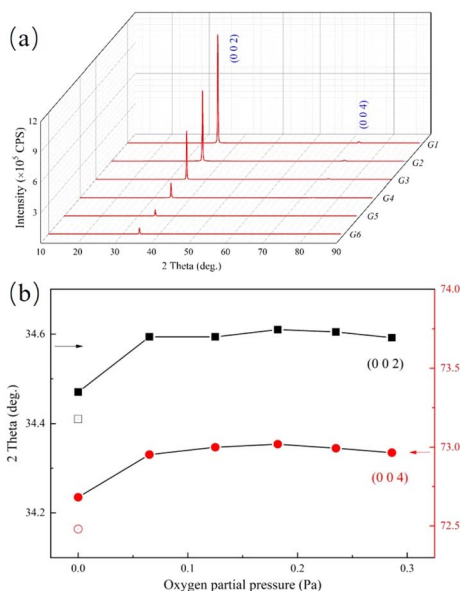


Fig. 1 (a) XRD patterns of GZO; (b) the (0 0 2) and (0 0 4) peak positions of GZO at different oxygen partial pressures were indicated by solid squares and circles, respectively. The hollow square and circle corresponded to (002) and (004) peak positions of ZnO.

respectively, which could be indexed to the (0 0 2) and (0 0 4) planes of the hexagonal wurtzite structure. There were only (0 0 2) and (0 0 4) peaks of the hexagonal wurtzite structure in each of the G1, G2 or G3 films, suggesting that the preferred orientation was along *c*-axis due to low surface free energy.¹⁰ As the oxygen partial pressure increased, the (1 0 0) peak, which belonged to the hexagonal wurtzite structure appeared. The ratio of the intensity of (0 0 2) peak to that of (1 0 0) peak, $R = I_{002}/I_{100}$, was calculated. The value of R_{ZnO} , derived from a completely random ZnO (JCPDS card no. 36-1451), was used as a reference. The R values for G4, G5 and G6 films were 558, 932 and 368, respectively, which were much larger than that of R_{ZnO} ($R_{\text{ZnO}} = 0.8$). The results proved that the preferred orientation was still along the *c*-axis, even though the oxygen partial pressure increased to 0.286 Pa (G6).

However, the intensities of both (0 0 2) and (0 0 4) peaks decreased with increasing oxygen partial pressure due to a reduction in film thickness and the crystallinity degradation.

In Fig. 1(b) the value of parameter-*c* for G1 film was 5.12 Å, which was much smaller than that of pure ZnO ($c = 5.21$ Å). This suggested that the Ga ions effectively doped into the lattice of ZnO matrix because of the smaller ionic radius of Ga and the shorter bond with oxygen.

As the oxygen partial pressure increasing to 0.065 Pa in G2 film, the positions of (0 0 2) and (0 0 4) peaks sharply increased, and the value of parameter-*c* decreased to 5.10 Å. Although the oxygen partial pressure continued to increase, this value nearly remained unchanged.

The decrease in the value of parameter-*c* might be due to the increasing concentration of V_{Zn} because of its lowest formation energy in an oxygen-rich condition.⁶ The high concentration of V_{Zn} would induce a great Coulomb force of attraction among the ions around it, which may cause crystal distortion and contraction of crystal lattice.

The strain values of GZO films could be calculated by the eqn (1):^{11,12}

$$\varepsilon = \frac{C_{\text{film}} - C_{\text{ZnO}}}{C_{\text{ZnO}}} \quad (1)$$

where the elastic constants C_{film} and C_{ZnO} corresponded to GZO film and ZnO. The stress value could be obtained from the eqn (2):

$$\sigma = \frac{2C_{13}^2 - C_{33}(C_{11} + C_{12})}{2C_{13}} \times \frac{C_{\text{film}} - C_{\text{ZnO}}}{C_{\text{ZnO}}} \quad (2)$$

where C_{11} , C_{12} , C_{13} , and C_{33} were the elastic constants of single crystal ZnO, whose values were 208.8, 114.7, 104.2 and 213.8, respectively.¹² The strain and stress of GZO films were shown in Table 2. It could be observed that the stress and strain of GZO increased under oxygen partial pressure.

Table 2 The strain and stress of GZO films deposited under the different oxygen partial pressures

	G1	G2	G3	G4	G5	G6
ε (%)	−1.68	−2.05	−2.05	−2.09	−2.07	−2.03
σ (GPa)	3.85	4.68	4.68	4.76	4.72	4.63



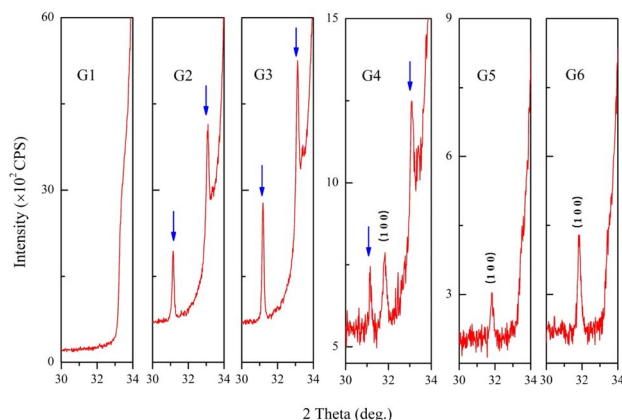


Fig. 2 The XRD patterns of GZO at the range of 30°–34°.

The impurity phases were analyzed by enlarging the XRD patterns. As shown in Fig. 2, there was no peak at the 2θ range of 30°–33° in G1, however, at the oxygen partial pressure of 0.065 Pa (G2), two new peaks appeared at 31.14° and 33.09°, which were indicated by arrows. The new peaks could be indexed to the orthorhombic structure of $\text{Ga}_2\text{Zn}_6\text{O}_9$ (ICSD #173602).¹³ The intensity of each new peak was much lower than that of (0 0 2) peak from GZO, which indicated that the amount of the impurity was very small.

When the oxygen partial pressure increased to 0.182 Pa (G4), another new diffraction peak appeared at 31.16°, and it

belonged to the (1 0 0) plane of GZO. The intensities of peaks belonging to $\text{Ga}_2\text{Zn}_6\text{O}_9$ gradually diminished and disappeared when the oxygen partial pressure was 0.235 Pa (G5) or higher. The impurity of $\text{Ga}_2\text{Zn}_6\text{O}_9$ appeared and disappeared during the increase of oxygen partial pressure, which implied that Ga ions might extract from the lattice of ZnO first and existed in the form of an amorphous state at higher oxygen partial pressure finally. This result suggested the effective doping of Ga_{Zn} in lattice of ZnO matrix reduced. The parameter- c was almost unchanged in oxygen-rich conditions due to a small trace of impurity. Moreover, the diffraction peaks from Ga, Ga_2O_3 or ZnGa_2O_4 were not detected.

The defects and chemical states of GZO films were characterized by XPS. $\text{Ga-2p}_{3/2}$ peaks were decomposed into two peaks at 1117.9 eV and 1118.6 eV, as shown in Fig. 3(a) and (b). The peak at 1117.9 eV was derived from the $\text{Ga}_{\text{Zn}}\text{-O}$ bonds, in which Ga atoms substituted for Zn atoms in ZnO matrix, while the peak at 1118.6 eV could be attributed to the Ga–O bonds due to GaO_x sub-oxides and oxides in intra-grain congregation or grain-boundary segregation.^{14–16} The intensity for the $\text{Ga-2p}_{3/2}$ peak at 1117.9 eV decreased while it increased for the $\text{Ga-2p}_{3/2}$ peak at 1118.6 eV. The results suggested that a number of Ga ions which replaced Zn in ZnO matrix were lost from the lattice of the matrix and might exist in the form of GaO_x sub-oxides with the increasing oxygen partial pressure.

The asymmetric peak observed in the O-1s spectrum of G1 film could be fitted by three types of peaks, as shown in Fig. 3(c). The lowest binding energy peak of O-1s, centered at 530.6 eV,

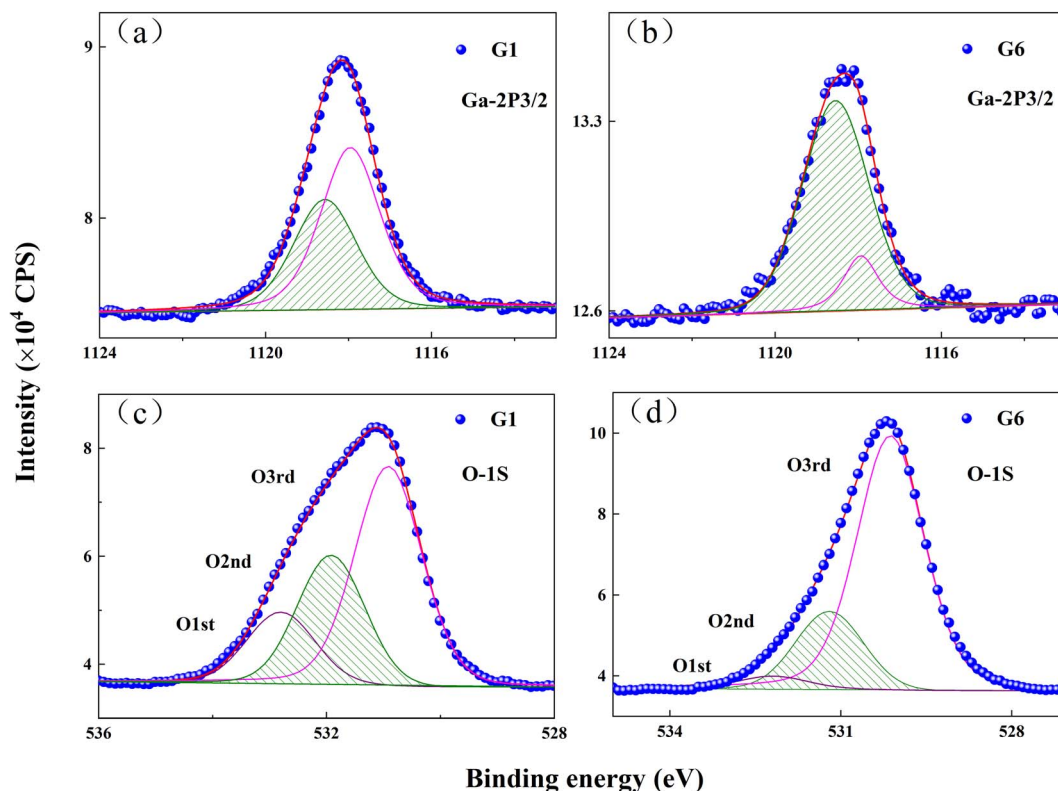


Fig. 3 XPS spectra of $\text{Ga-2p}_{3/2}$ peaks for: (a) G1 film; (b) G6 film, and O-1s peaks for: (c) G1 film; (d) G6 film.

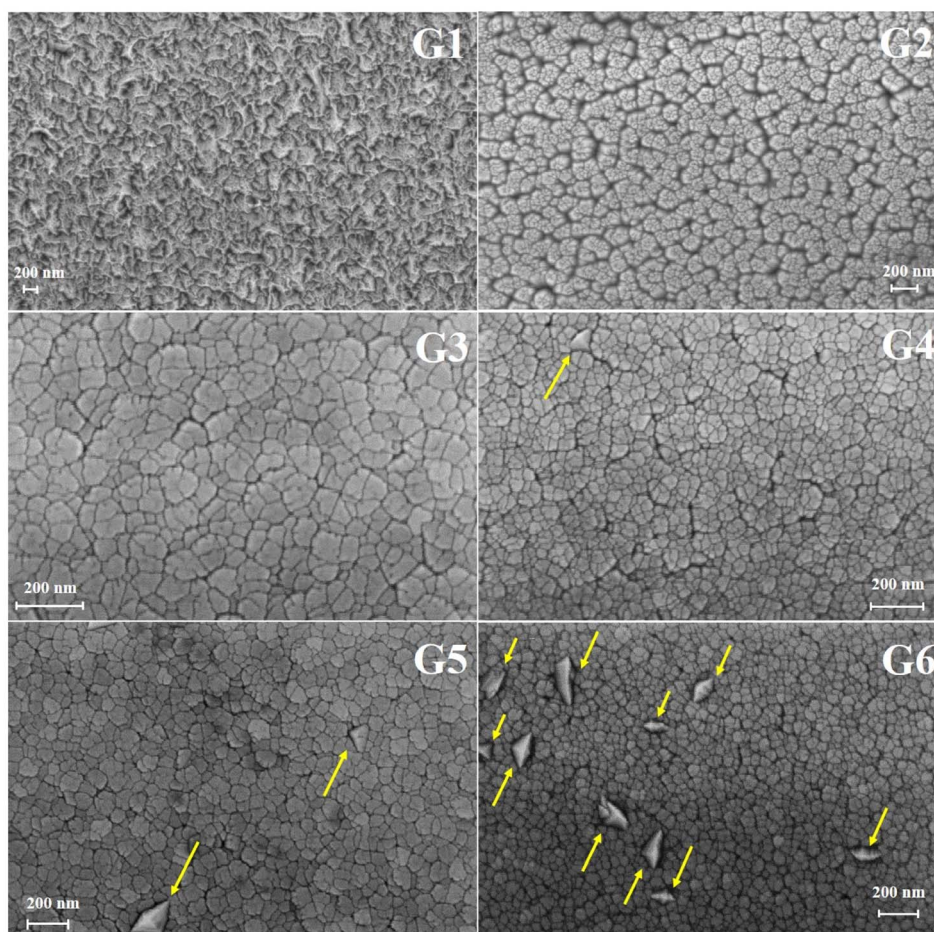


Fig. 4 SEM images of GZO films deposited at various oxygen partial pressures.

was marked as O_{3rd} and derived from the bonds between O^{2-} and metal ions (Zn^{2+} and Ga^{3+}).^{17,18} The middle peak at 531.3 eV (O_{2nd}) corresponded to oxygen vacancies in the G1 film.^{19,20} The highest binding energy at 532.2 eV was marked as O_{1st} , which was related to oxygen near surface, such as O–H bonds.^{21,22} In comparison with the G1 film, the O_{1st} , O_{2nd} and O_{3rd} peaks of G6 film were located at 530.2 eV, 531.3 eV and 532.3 eV in Fig. 3(d). The intensity of the O_{2nd} peak was proportional to the concentration of oxygen vacancies.²³ Peak intensity ratio of O_{2nd}/O_{total} was 26.7% for G1 film, but it decreased to 21% for G6 film, which indicated that the V_O concentration in GZO film obviously decreased at the oxygen partial pressure.

The effect of oxygen partial pressure on the morphology of the GZO films was analyzed by using SEM. As shown in Fig. 4, the surface of the G1 film appeared compact and crack-free. In contrast, G2 film exhibited wide and deep boundaries, which became narrower in G3 film. The G4 film showed large grains with a tetrahedral shape, as indicated by arrows, and the number of these grains increased with increasing oxygen partial pressure.

In general, the compact surface of G1 film turned into grains with a lot of boundaries in oxygen-rich conditions. The results suggested the nucleation and growth process during the

formation of GZO films might be affected by oxygen, which reduced the island size in the coalescence stage.^{24,25} By comparison, it has been observed that the grain size of ZnO:Mg film decreased with the increasing oxygen partial pressure.²⁶

Additionally, the tetrahedral grains could be caused by abnormal grain growth. In the oxygen-rich conditions, the

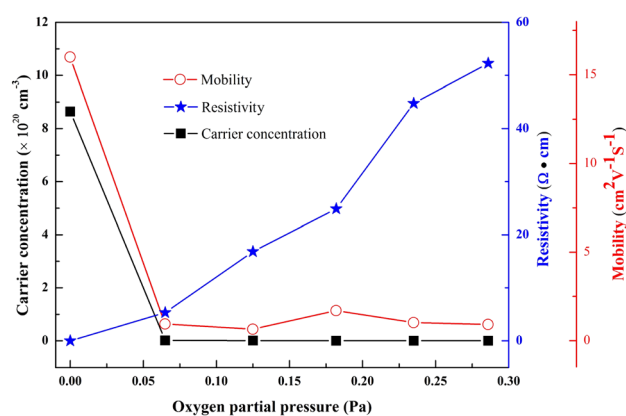


Fig. 5 Resistivity, carrier concentration and Hall mobility of GZO films under various oxygen partial pressures.



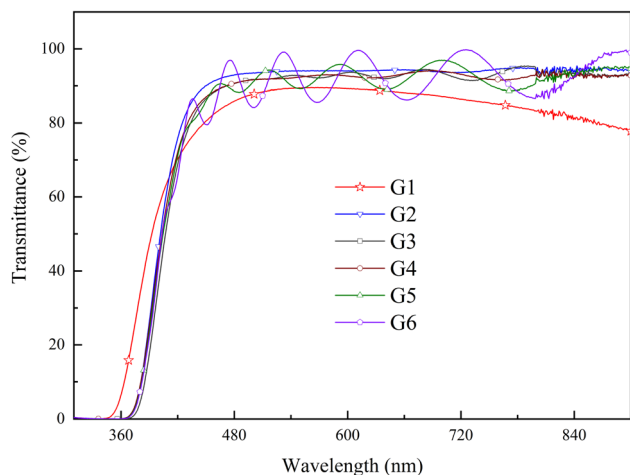


Fig. 6 Transmission spectra of GZO films in the region of 300–900 nm.

mobility of boundary migration was limited, but some grain boundaries might have overcome the pinning effect and grown into larger sizes compared to other boundaries that were still pinned.²⁷

3.1 Hall

The Hall effect measurement was conducted to analyze the electrical properties of the GZO films. In Fig. 5 the carrier concentration and resistivity were $8.6 \times 10^{20} \text{ cm}^{-3}$ and $4.5 \times 10^{-4} \Omega \text{ cm}$ for the G1 film. In contrast, carrier concentration of

the G2 film, which was deposited at an oxygen partial pressure of 0.065 Pa, decreased abruptly to $1.2 \times 10^{18} \text{ cm}^{-3}$ and the resistivity increased to $5.3 \Omega \text{ cm}$. With increasing oxygen partial pressure, the carrier concentration gradually reduced ($5.7 \times 10^{17} \text{ cm}^{-3}$, $1.5 \times 10^{17} \text{ cm}^{-3}$ and $1.4 \times 10^{17} \text{ cm}^{-3}$ for G3, G4 and G5 films), and the resistivity rose slowly ($16.8 \Omega \text{ cm}$, $24.9 \Omega \text{ cm}$ and $44.8 \Omega \text{ cm}$ for G3, G4 and G5 films). At the highest oxygen partial pressure of 0.286 Pa (G6), the concentration and resistivity reached values of $1.3 \times 10^{17} \text{ cm}^{-3}$ and $52.3 \Omega \text{ cm}$, respectively. The same changes were also observed for mobility, which was as high as $16 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$ for the G1 film, but it decreased to $0.9 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$ for the G6 film. The decrease of mobility might mainly be caused by the increase of electron scattering induced by grain boundary.

The high n-type carrier concentration (n_{total}) for the G1 film was due to the high concentrations of Ga_{Zn} and V_{O} donor defects. The carrier concentration was reduced mainly due to the decrease of V_{O} concentration and the increase of V_{Zn} concentration, which acted as compensating intrinsic defects.⁵

3.2 Transmittance

Optical transmittance spectra from 300 to 900 nm were analyzed by UV-vis spectrophotometer in Fig. 6. All of the spectra exhibited sharp absorption edges, and the average optical transmittance of each GZO film deposited under oxygen partial pressure was more than 90% for the wavelength above 450 nm. The optical transmittance of G1 film gradually decreased as the wavelength increased in the near-infrared (NIR) region due to absorption by the free carrier plasm

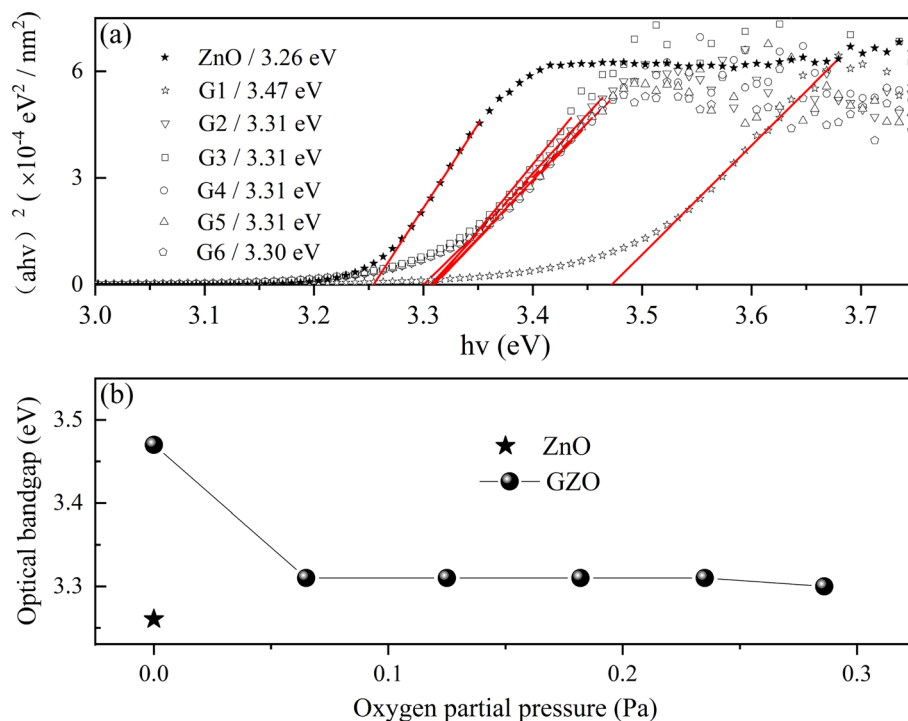


Fig. 7 The (a) and (b) showed the optical bandgap of GZO (ball) varied with the O_2 partial pressure and the value of ZnO was indicated by the symbol of star.



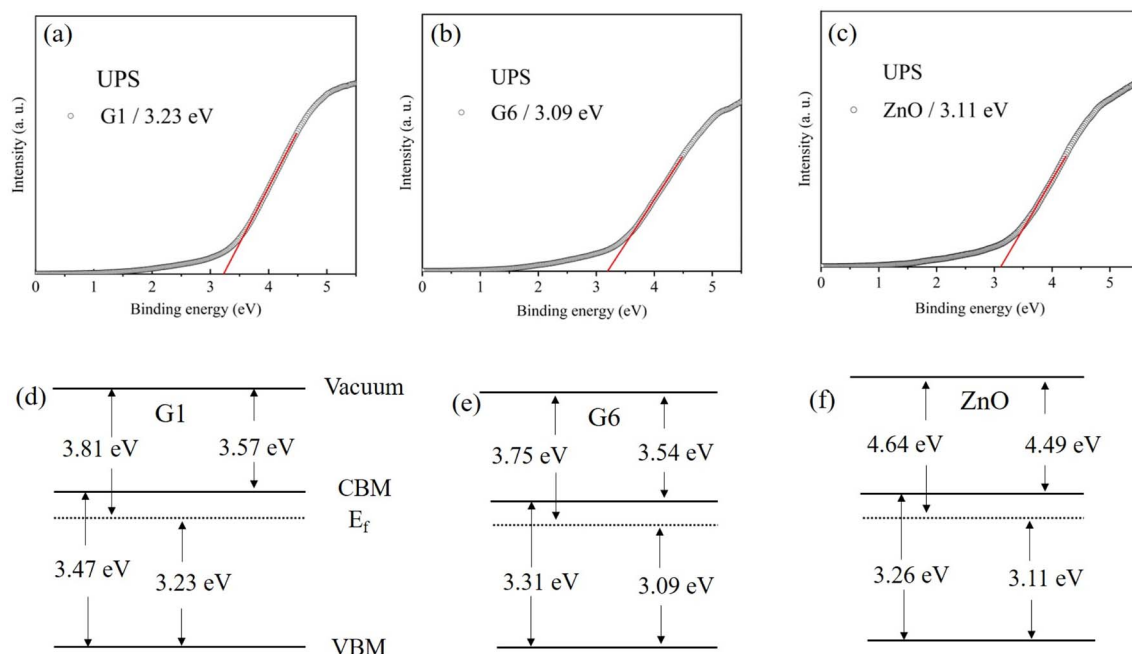


Fig. 8 (a)–(c) were UPS spectra which correspond to G1, G6 and ZnO films. (d)–(f) Were sketches of band structures which correspond to G1, G6 and ZnO films.

resonance frequency.²⁸ The absorption edge of the G1 spectrum shifted to longer wavelength in oxygen-rich conditions.

Fig. 7(a) and (b) showed the variation of the optical bandgap with increasing oxygen partial pressures. The bandgap of the G1 film was 3.47 eV and decreased rapidly to 3.31 eV under an oxygen partial pressure of 0.065 Pa (G2). The optical bandgaps then remained almost constant at 3.31 eV with increasing oxygen partial pressure, which were larger than that of the pure ZnO film (3.26 eV).

The optical bandgap of the G1 film was the largest due to the Burstein–Moss (B–M) effect, which described the blue-shift of the optical bandgap energy. The relationship between the broadening of the optical band gap (ΔE_g) and carrier concentration (n) could be expressed by eqn (3):^{3,29}

$$\Delta E_g = \left(\frac{h^2}{2m^*} \right) \left(\frac{3n}{\pi} \right)^{2/3} \quad (3)$$

where h was the Planck's constant and m^* was the effective mass of electron. The ΔE_g was in positive relationship with carrier concentration. The concentration of the electrons reduced from the $8.6 \times 10^{20} \text{ cm}^{-3}$ for the G1 film to $1.3 \times 10^{17} \text{ cm}^{-3}$ at an oxygen partial pressure of 0.286 Pa (G6 film). As the result, the optical band gap significantly decreased according to eqn (3).

The UPS spectra for G1, G6, and ZnO films were shown in Fig. 8(a)–(c), respectively. The error tolerances in our analyses were ± 0.15 eV. The VBM was determined by the cross point at the linear extrapolation of the edge of UPS and the baseline (X-axis). The band structures of G1, G6 and ZnO were shown by Fig. 8(d)–(f), respectively, by combining them with the optical band gap. The electron affinity of G1 was 3.57 eV, while it was 3.54 eV for G6. The result suggested that oxygen partial pressure

had almost no effect on the electron affinity, which was a key parameter in band alignment of device consisting with multi-layer films. For comparison, the electron affinity of ZnO was as high as 4.49 eV, which was much larger than G6, although both ZnO and G6 were insulated. The experimental results showed that the electron affinity of ZnO could be effectively tuned by doping Ga, but electron affinity of GZO had almost not been affected by oxygen partial pressure.

4. Conclusions

In summary, the GZO films were deposited by using RF sputtering method. The effects of oxygen partial pressure on structure and performance of GZO films have been investigated. The value of parameter- c decreased at an oxygen partial pressure of 0.065 Pa and then almost remained unchanged until the pressure increased to 0.286 Pa. It was the first time that the impurity of $\text{Ga}_2\text{Zn}_6\text{O}_9$ appeared first and disappeared finally with the increase in oxygen partial pressure.

The opto-electrical properties of GZO films deposited at different oxygen partial pressures were measured. The resistivity increased from $4.5 \times 10^{-4} \Omega \text{ cm}$ to $52.3 \Omega \text{ cm}$ when the oxygen partial pressure increased from 0 to 0.286 Pa, mainly due to the decrease of V_O concentration and the increase of V_{Zn} concentration, which acted as compensating intrinsic defects. Meanwhile, the optical bandgap of GZO film decreased with the increase in oxygen partial pressure from 3.47 eV to about 3.30 eV. The results showed that the conductivity and optical bandgap of GZO film were easily affected by oxygen partial pressure. Additionally, the electron affinities were 3.57 eV and 3.54 eV for G1 and G6, respectively, and it was proved that the



oxygen partial pressure had almost no effect on the electron affinity of GZO film in oxygen-rich condition. For comparison, the electron affinity of ZnO film was 4.49 eV, which suggested that Ga played an important role in tuning electron affinity of ZnO film.

Conflicts of interest

There are no conflicts to declare.

Acknowledgements

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