Check for updates

# Effects of functional groups in iron porphyrin on the mechanism and activity of oxygen reduction reaction†

Tran Phuong Dung,[ab] Viorel Chihaia[c] and Do Ngoc Son [ID] *[de]

The activity of the oxygen reduction reaction (ORR) on the cathode is one of the dominant factors in the performance of proton exchange membrane fuel cells. Iron porphyrin has low cost, environmental benignity, and maximum efficiency of metal usage. Therefore, this material can be a promising single-atomic metal dispersion catalyst for fuel cell cathodes. The variation of functional groups was proven to effectively modify the activity of the ORR on the iron porphyrin. However, the influences of functional groups on the mechanisms of the ORR remain ambiguous. This work paid attention to the substitution of carboxyl ($-COOH$), methyl ($-CH_3$), and amino ($-NH_2$) functional groups at the *meso* positions of the porphyrin ring. By using van der Waals density functional theory (vdW-DF) calculations, we found that the ORR mechanisms can follow the associative and dissociative pathways, respectively. The Gibbs free energy diagrams revealed that the rate-limiting step occurs at the second hydrogenation step for the first pathway and the $O_2$ dissociation step for the second pathway for all considered functional groups. The thermodynamic energy barrier at the rate-limiting step was found to be in the following order: porphyrin$-(CH_3)_4$ < porphyrin$-(NH_2)_4$ < original porphyrin < porphyrin$-(COOH)_4$ for the associative mechanism and porphyrin$-(NH_2)_4$ < porphyrin$-(CH_3)_4$ < porphyrin$-(COOH)_4$ < original porphyrin for the dissociative pathway. The findings suggested that porphyrin$-(CH_3)_4$ and porphyrin$-(NH_2)_4$ should be the best choices among the considered substrates for the oxygen reduction reaction. Furthermore, the interaction between the ORR intermediates and the substrates was attributed to the resonance of the $d_{z^2}$, $d_{xz}$, and $d_{yz}$ components of the Fe d orbital and the C and N p orbitals of the substrates with the p orbitals of the oxygen atoms in the intermediates. Finally, the nature of the interaction between the adsorbent and adsorbate was charge transfer.

## 1. Introduction

Hydrogen gas is a clean and renewable energy source for proton exchange membrane fuel cells (PEMFCs).[1] The electrochemical energy conversion of the hydrogen gas easily and silently happens on the anode of the PEMFCs. However, the oxygen reduction reaction on the cathode involves many intermediate steps and energy consumption. The employment of an expensive Pt catalyst for the ORR at the cathode weakens the

*aDepartment of Chemistry, University of Science, Vietnam National University, Ho Chi Minh City, Vietnam*

*bDepartment of Chemistry, Ho Chi Minh City University of Education, Ho Chi Minh City, Vietnam*

*cInstitute of Physical Chemistry "Ilie Murgulescu" of the Romanian Academy, Splaiul Independentei 202, Sector 6, 060021 Bucharest, Romania*

*dHo Chi Minh City University of Technology (HCMUT), 268 Ly Thuong Kiet Street, District 10, Ho Chi Minh City, Vietnam. E-mail: dnson@hcmut.edu.vn*

*eVietnam National University Ho Chi Minh City, Linh Trung Ward, Ho Chi Minh City, Vietnam*

† Electronic supplementary information (ESI) available. See DOI: https://doi.org/10.1039/d2ra08007a

competitiveness of the PEMFCs in the energy market. Besides, the presence of contaminant gases such as $CO_x$, $NO_x$, and $SO_x$ reduces the active sites on the catalytic surface of the cathode; and hence, decreases the durability and catalytic performance of the Pt electrode.[2] Many strategies have been used to improve the catalytic efficiency and cut down the precious metal content: (1) alloy Pt with earth-abundant and low-cost transition metals,[3,4] (2) disperse Pt-based alloys on graphene nanoplatelets,[5] (3) create the Pt-skin surface of alloys,[3] and (4) synthesize the nanoparticles of Pt-based alloys with carbon supports.[6,7] The literature also showed that annealing at high temperatures enhanced the ORR performance and the stability of the Pt-based alloys.[4,5,8] Notably, the Pt-free alloys of non-precious metals have been investigated for the ORR, and found that, they not only offered higher stability than the Pt-based alloys, but also improved the ORR performance even better than that of commercial Pt/C nanoparticles.[9–12] However, the usage of metals faces the problems of electrode corrosion during the operation of the PEMFCs and environmental unfriendliness.[13]

Recently, much effort has been made to search for low-cost, high-performance, and low-metal loading catalysts. Single-

atomic metal catalysts of non-precious transition metals (Fe, Co, Mn, Ni, *etc.*) and nitrogen-doped carbon materials exhibited promising efficiency and stability for the ORR.[8,13–15] Particularly, the complexes of transition metals and porphyrin known as metalloporphyrins have attracted much attention due to the high flexibility of tuning their geometrical and electronic structures.[16] The literature showed that substituting functional groups such as tetracarboxyphenyl, tetrapyridyl, and *N*-methyl-4-pyridyl could modify the ORR mechanism and activity of metalloporphyrins.[17–19] Experimentally, the catalytic activity and mechanism were often studied *via* the cathodic current density, the redox potential of the catalyst, and the half-wave potential value of the cathodic peak. In particular, the complete reduction of $O_2$ *via* a 4-electron pathway leading to high energy conversion efficiency was promoted by the formation of dimers such as face-to-face cobalt porphyrins[20] and linear dimer of iron porphyrins.[21] Previous studies have shown that the size of alkyl groups at *meso* positions of cobalt porphyrin shifted the half-wave potential for the ORR to a more negative value and decreased the number of electrons involved in the ORR from 3.8 *e* to 2.7 *e*.[20,22] The other functional groups as pyridyl and carboxyphenyl could facilitate the ORR with the 4-electron mechanism due to more protons available to reduce the $O_2$ intermediate on iron porphyrin.[17,18] Also, the modification of the functional groups changed the electrochemical properties of metalloporphyrins and influenced the oxygen reduction ability of the metals in the core region of the porphyrin structure.[23] Besides, the electron-withdrawing groups in iron tetraphenyl porphyrin modified the relationship between reduction rate and overpotential due to different rate-determining steps.[24] However, no works are available to clarify the influences of methyl, amino, and carboxyl functional groups on the ORR mechanism of iron porphyrin. Therefore, this paper is devoted to solving the mentioned topic using density functional theory calculations and the thermodynamic model.

The present work searched for possible reaction intermediates on the porphyrin substrates with the methyl, amino, and carboxyl functional groups at the *meso* positions of the porphyrin ring. We then explored the structural and electronic properties of the obtained intermediates. Finally, we proposed the ORR mechanisms, calculated Gibbs free energy for the intermediate steps, and built and analyzed the Gibbs free energy diagrams to understand the mechanisms. The obtained results were compared to those available in the literature.

## 2. Computational details

This study was carried out using the van der Waals density functional theory (vdW-DF) calculations within the revised version of the generalized gradient approximation of Perdew–Burke–Ernzerhof (GGA-revPBE) for the exchange–correlation energy by the vienna *ab Initio* simulation package (VASP).[25] The projector-augmented wave method[26] was applied to smoothen the core potential with a plane-wave cutoff energy of 450 eV. The *k*-point mesh of $3 \times 3 \times 3$ was generated by the Monkhorst and Pack method[27] for the integration in the first Brillouin zone. This *k*-point mesh sample has been tested to ensure the

convergence of total energy. In the present work, Gaussian smearing with the width of 0.1 and 0.01 eV was used to accelerate the geometric optimization and for the calculation of the electronic density of states, respectively. The dipole correction along the surface normal of the porphyrin structure was included for the periodic supercells.

The unit cells must be large enough to contain the porphyrin structures with the different functional groups and guarantee the vacuum space of at least 10 Å in the *xyz* plane and 13 Å along the surface normal. Fig. 1 shows the structure of the iron porphyrin (FePor), *meso*-tetramethyl iron-porphyrin (FeTMP, porphyrin–$(CH_3)_4$), *meso*-tetraamino iron-porphyrin (FeTAP, porphyrin–$(NH_2)_4$), and *meso*-tetracarboxyl iron-porphyrin (FeTCP, porphyrin–$(COOH)_4$). All the atomic positions of the porphyrins and ORR intermediates were allowed freely moving during the geometric optimization.

The stability of the substrates can be determined *via* their formation energy, which is the difference between the total energy of iron-porphyrin with that of the isolated iron atom and the isolated porphyrin without iron. The formation energy of FePor, FeTMP, FeTAP, and FeTCP was −9.02, −8.50, −6.72, and −8.99 eV, respectively. The negative formation energy of the substrates implies that the incorporation of the iron atom into the porphyrin is thermodynamically favorable.

The total oxygen reduction reaction generates water as the following equation:

$$O_2 + 4(H^+ + e^-) \rightarrow 2H_2O \qquad (1)$$
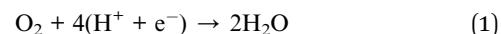
However, it can proceed *via* many intermediate steps and involve many reaction intermediates.

The adsorption energy of each intermediate (A) on substrate (B) was calculated by the formula:

$$E_a = E_{AB} - [E_A + E_B] \qquad (2)$$

where $E_{AB}$, $E_A$, and $E_B$ are the total energy of the optimized substrate-intermediate complex (AB), the isolated intermediate (A), and the clean substrate (B), respectively.

The Gibbs free energy $\Delta G$ for each intermediate step was determined *via* the proton and electron exchange model.[28–30] The energy combining a proton and electron was equivalent to the energy of $\frac{1}{2}H_2$ in the gas phase at the potential of the reversible hydrogen electrode, $H^+ + e^- = \frac{1}{2}H_2$. In the standard condition (pH = 0, $p$ = 1 bar, $T$ = 298 K),
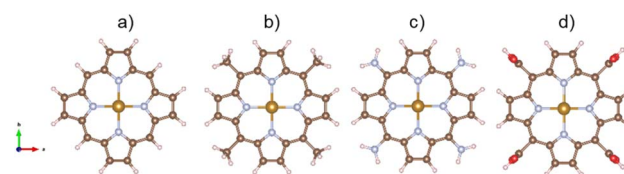


**Fig. 1** . The optimized structures of iron porphyrin (a), *meso*-tetramethyl iron-porphyrin (b), *meso*-tetraamino iron-porphyrin (c), and *meso*-tetracarboxyl iron-porphyrin (d). Fe (yellow), C (brown), N (cyan), H (pink), O (red).

$$\Delta G = \Delta E + \Delta ZPE - T\Delta S - ne U \qquad (3)$$

where $\Delta E$, $\Delta ZPE$, and $\Delta S$ are the reaction energy, the zero-point energy correction, and the entropy change of the intermediate step, respectively. $\Delta E$ and $\Delta ZPE$ were determined from the results of DFT calculations, and $T\Delta S$ was taken from ref. 30. $U$ is the cathode potential relative to the reversible hydrogen electrode at the same temperature. $n$ is the number of electrons in each reaction step. All the Gibbs free energies were calculated relative to the free energy of water as the final product of ORR. The Gibbs free energy of the oxygen molecule was corrected by the experimental free energy of the water-splitting reaction (2.46 eV).

## 3. Results and discussion

### 3.1. Structural properties

The ORR begins with the adsorption of $O_2$ molecules on the substrate. The hydrogenation steps happen by continuously loading hydrogen atoms onto the previously adsorbed intermediates. The possible adsorption positions on the porphyrin substrates are the top of the Fe, C, and N atoms, over the Fe–N, N–C, and C–C bridges, and the hollow sites of pyrrole and Fe–N–C rings. After performing the structural relaxation by the DFT calculations, we obtained the optimized geometric structures of the substrate-intermediate systems. According to definition (2), the adsorption structure with negative adsorption energy should be favorable. The more negative adsorption energy, the more stable level is. Fig. 2a–h displayed the most stable configuration of each intermediate on the porphyrin substrate, which has the most negative adsorption energy, as shown in Table 1.

**3.1.1 The associative scenario.** The $O_2^*$ intermediate adsorbs most favorably on the iron atom with the end-on configuration (Fig. 2a). The asterisk (*) denotes the adsorption state of the intermediate on the substrate. It should be noted that the side-on configuration of the $O_2^*$ intermediate on the iron atom was also obtained. However, it is less stable than the end-on one because its adsorption energy is more positive by about 0.293 eV. Besides, the adsorption energy of $O_2$ on FePor in the present work agrees with that of ($-15.6 \pm 0.2$ kcal mol$^{-1}$ $\approx$ $-0.68 \pm 0.01$ eV) obtained by the experiment.[31] The first hydrogenation product of the associative scenario, HOO*, also adsorbs with the end-on configuration (Fig. 2b) and enhances the adsorption strength in comparison to the $O_2^*$ intermediate, see Table 1. It is possible that HOO* dissociates into O* + HO* to become a more stable intermediate. The next hydrogenation step generates HOOH* (Fig. 2c) and O* + $H_2O$* (Fig. 2d), where O* + $H_2O$* is more stable than HOOH*. Besides, HOOH* is less favorable than HO* + HO* because the magnitude of its adsorption energy is significantly lower than that of the latter. Therefore, the HOOH* intermediate is possible to dissociate into HO* + HO*. The third hydrogenation product is HO* + $H_2O$* (Fig. 2e), which has the most negative adsorption energy among the ORR intermediates. It must be noted that the O* + HO*, HO* + HO*, and HO* + $H_2O$* intermediates occur in both associative and dissociative scenarios. Therefore, we have to

discuss these intermediates in the present paragraph for the completion of the description. Updating the last (fourth) hydrogen atom onto HO* of HO* + $H_2O$* can generate two water molecules, 2$H_2O$*, the by-product of the ORR. This final product slightly adsorbs on the surface of porphyrin substrates with the least negative adsorption energy.

**3.1.2 The dissociative scenario.** The adsorption energy magnitude of $O_2^*$ is smaller than that of O* + O*. Therefore, two oxygen atoms can separately adsorb at two different sites in a more stable state. One oxygen atom is on the top of the Fe atom, and the other is above the C–C bond of the pyrrole ring (Fig. 2f). The first and second hydrogenation steps of the O* + O* intermediate generate O* + HO* and HO*+ HO* (Fig. 2g and h), respectively. The third and fourth hydrogenation steps follow the same way as the third and fourth steps of the associative scenario accordingly.

We obtained stable structures for the ORR intermediates on the FePor, FeTMP, FeTAP, and FeTCP substrates. We found that the adsorption site and configuration of each intermediate are similar for these substrates. Table 1 also shows that the adsorption strength of each intermediate is highest on FeTAP. Furthermore, the adsorption energy is negative, while the zero-point energy is positive. The stability of each intermediate depends on the competition between the adsorption energy and the zero-point energy. As shown in Table 1, the intermediates are stable except for HOOH* on FePor, and 2$H_2O$* on all the substrates. Remarkedly, the 2$H_2O$* intermediate exhibits a much higher zero-point energy than the absolute value of its adsorption energy. Therefore, it perhaps leaves the adsorption surface after completing the oxygen reduction reaction. For the intermediates that have the same number of hydrogen atoms ($O_2^*$ *vs.* O* + O*, HOO* *vs.* O* + HO*, and HOOH* *vs.* HO* + HO*), the intermediate with atomic oxygens will adsorb stronger than the intermediate with the molecular oxygen on the substrate surface. The HO* + $H_2O$* intermediate is the strongest adsorption intermediate compared to the others. This finding is in good agreement with the previous publication that the HO* intermediate covered the Fe site in a wide range of electrode potentials.[32]

One oxygen atom of each intermediate always adsorbs on the iron site (Fig. 2) with the O–Fe bond distance of about 1.6 to 1.9 Å except for that of about 2.5 Å for 2$H_2O$*, while the O–O distance increases as the number of hydrogen atoms in the intermediates increases (Table S1 in ESI†). The second oxygen atom in the dissociative adsorption scenario is always at the boundary of the substrates. Particularly, each intermediate in the dissociative state shows a significantly large O–O distance compared to that of the associative state. The more the negative adsorption energy, the more the elongated O–O bond length of the oxygen atoms.

### 3.2 Electronic properties

The nature of the interaction between the ORR intermediates and the porphyrin substrates can be explained *via* the electronic structure properties, *i.e.*, the Bader point charge, the charge density difference, and the electronic density of states.
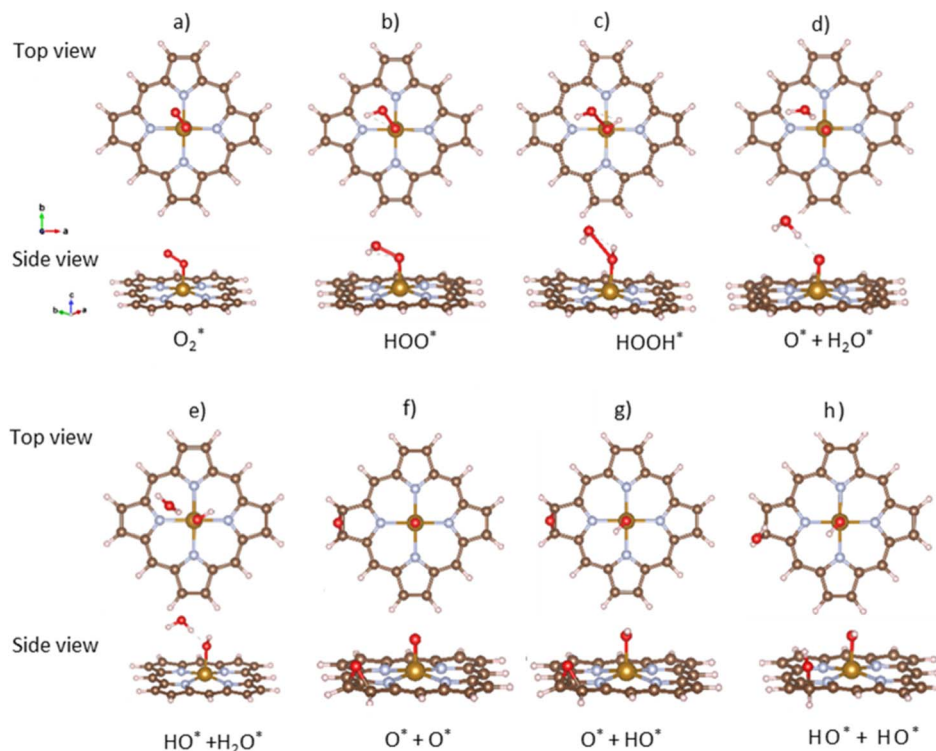
**Fig. 2** The most favorable adsorption configuration of the ORR intermediates. The top view (above) and the side view (below). Fe (yellow), C (brown), N (light blue), O (red), and H (pink).

To make the discussion simple, we divided the porphyrin structures into the core region and the functional group. According to this partition, the functional groups include the H atoms, $-CH_3$, $-NH_2$, and $-COOH$ for the porphyrin substrates. The core region includes the Fe, C, and N atoms of the original iron porphyrin, see Scheme 1. The Bader charge has been calculated by subtracting the charge of the neutral atoms from the charge of the corresponding atoms in the substrate. The Bader charge of the clean substrates in Table 2 shows that, for the different functional groups, the Fe and C atoms in the core region of the porphyrin ring always donate, while the N atoms always accumulate, the negative charge ($e^-$) compared to their neutral state. As a result, the core region of clean FePor and FeTMP gains, while that of clean FeTAP and FeTCP loses a negative charge. The functional groups have to compensate for the charge variation of the core region so that the substrates become neutral. Particularly, while the H and TMP functional groups donate, the TAP and TCP groups gain the charge. This finding implies that the H and TMP groups exhibit the charge-donating to, while the TAP and TCP show charge-withdrawing nature from, the core region of the porphyrin. On the other hand, according to the classical viewpoint of chemistry, the influences of functional groups on the charge distribution to be defined as the electron-donating and -withdrawing are based on the charge gain and loss of the reaction center, the Fe atom in this study. For the substitution of the functional groups, we found that the charge loss of the Fe atom decreases for TMP and increases for TAP and TCP compared to FePor. Therefore, TMP reveals electron-donating while TAP and TCP show electron-withdrawing properties. Besides, the Bader charge of TAP

**Table 1** The adsorption energy (eV) of the ORR intermediates on iron porphyrin (FePor), tetra-methyl porphyrin (FeTMP), tetra-amino porphyrin (FeTAP), tetra-carboxyl porphyrin (FeTCP). Zero-point energy (eV) in parenthesis

| Intermediates | FePor | FeTMP | FeTAP | FeTCP |
|---|---|---|---|---|
| $O_2^*$ | −0.704 (0.141) | −0.734 (0.142) | −0.997 (0.143) | −0.675 (0.143) |
| HOO* | −1.513 (0.408) | −1.544 (0.413) | −1.859 (0.403) | −1.516 (0.403) |
| HOOH* | −0.487 (0.606) | −0.733 (0.611) | −0.960 (0.617) | −0.661 (0.617) |
| O* + H$_2$O* | −2.389 (0.691) | −2.466 (0.693) | −2.856 (0.689) | −2.372 (0.689) |
| HO* + H$_2$O* | −2.808 (0.954) | −2.852 (0.949) | −3.281 (0.982) | −2.797 (0.946) |
| 2H$_2$O* | −0.481 (1.258) | −0.445 (1.250) | −0.658 (1.262) | −0.453 (1.254) |
| O* + O* | −1.291 (0.174) | −1.287 (0.172) | −1.492 (0.170) | −1.250 (0.174) |
| O* + HO* | −2.221 (0.414) | −2.211 (0.436) | −2.490 (0.453) | −2.222 (0.458) |
| HO* + HO* | −1.414 (0.715) | −1.460 (0.639) | −1.943 (0.626) | −1.475 (0.717) |
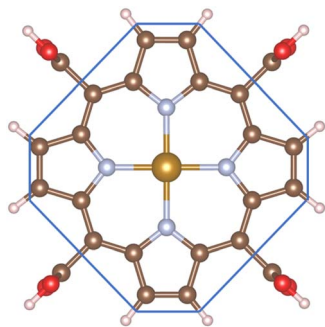
**Scheme 1** The core region is inside, and the functional group is outside the blue polygon boundary.

shows a large negative charge accumulation due to the charge-gaining nature of the N atoms of the TAP group. Furthermore, the charge loss of the Fe metal center decreases in the following order FeTAP > FeTCP > FePor > FeTMP.

Upon the adsorption of the intermediates, the Fe, C, and N atoms retain the charge donation and accumulation properties as that of the clean substrates. However, the detailed variation for each atom of the core region will be different. We found that all the intermediates significantly gain the charge from the substrates except for the final product $2H_2O^*$ donating the charge to the substrates, which indicates that the nature of the interaction between each intermediate and the substrates is the charge exchange. The charge difference, $\delta C$, for the Fe, C, and N atoms in the core region and the functional group of the FePor, FeTMP, FeTAP, and FeTCP substrates shows the minus or plus signs depending on the various intermediates. Even though the core region of each substrate with the adsorption of the intermediates, except for $2H_2O^*$ on FeTCP, always loses the charge relative to the clean substrate. The significant charge loss of the core region was found to mainly stem from the donation of the Fe and C atoms of the porphyrin ring.

Table S2 in ESI,† shows the charge of the oxygen atoms of the intermediates alone. We found that the oxygen atoms of all the intermediates are the centers of the charge gain that also exhibit in the charge density difference (Fig. 3). We also can see the charge loss clouds occurring near the Fe and C atoms. The total accumulating charge of the two oxygen atoms for each intermediate was presented in Fig. 4. The higher the order of hydrogenation, $i.e.$, from blocks 1 to 6 for the associative and 7 to 9 for the dissociative scenario, the higher the total charge gain of the oxygen atoms is. It should be noted that the intermediates in blocks 4, 5, and 6 are considered as those that belong to both the associative and dissociative adsorption pathways. By comparing the molecular state with the atomic state of the intermediates, blocks 1 with 7, 2 with 8, and 3 with 9, we found that the charge gain of the oxygen atoms of the intermediates is more significant in the dissociative adsorption than the associative one. The total charge difference between the pairs of these blocks is about 1.0 to 1.2, 0.7 to 0.8, and 0.4 to 0.6 $e^-$, respectively. Physically, when two oxygen atoms of the intermediates gain more charge, the coulomb-repulsive force between them increases. This force pushed two oxygen atoms
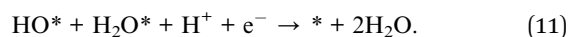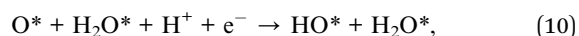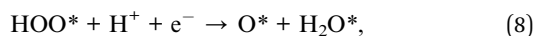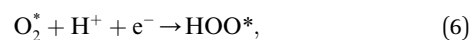
away to form more stable states. Therefore, the dissociative intermediates have more negative adsorption energy and longer O–O bond distance compared to the associative ones with the same number of hydrogen atoms, as shown in Table S1 in ESI.†

A deeper understanding of the nature of the interaction between the ORR intermediates and the substrates can be exposed $via$ the analysis of the electronic density of states (DOS). Fig. 5 (also Fig. S1 and S2 in ESI†) describes the DOS of the substrate-intermediate systems for the intermediates in the associative and dissociative scenarios. The resonance or overlap of the DOS peaks between the adsorbate and the adsorbent identifies the nature of their interaction, which induces the charge transfer from the substrates to the intermediates as mentioned in the Bader charge analysis of Table 2. In Fig. 5 (and Fig. S1 and S2†), we only drew the most significant orbitals near the Fermi level. Also, the Fe $d_{xz}$ and Fe $d_{yz}$ orbitals are identical in all plots. Therefore, we plotted one of them, $i.e.$, the Fe $d_{xz}$ orbital. Fig. 5, S1, and S2† show that the p states ($p_z$ more often than $p_x$ and $p_y$) of the oxygen atoms of the intermediates around the Fermi level participate in the interaction with the substrates. Simultaneously, the z-components of the Fe d orbital, $i.e.$, $d_{xz}$ ($d_{yz}$) and $d_{z^2}$ of the substrates are the main contributors to the interaction with the ORR intermediates. In several cases, for example, $O_2^*$ on FeTCP, HOO* on FeTAP, HOOH* on FeTMP, and the others, the C $p_z$ and N $p_z$ orbitals also make a significant contribution. Generally speaking, the z-components of the Fe d orbitals and the C and N p orbitals of the substrates with the O p orbitals of the intermediates around the Fermi level are attributed to the physical nature of the adsorbate–adsorbent interaction for the ORR on the porphyrin substrate. The DOS was found to support the Bader charge analysis.

### 3.3. Mechanisms of oxygen reduction reaction

Based on the obtained ORR intermediates, the total oxygen reduction reaction (1) can proceed in two scenarios, $i.e.$, the dissociative and associative pathways, as described in Fig. 6.

The associative mechanism starts with the molecular adsorption of $O_2$ on the substrates as follows:[29,33]

$$O_2 + * \rightarrow O_2^*, \tag{5}$$

$$O_2^* + H^+ + e^- \rightarrow HOO^*, \tag{6}$$

$$HOO^* + H^+ + e^- \rightarrow HOOH^*, \tag{7}$$

$$HOO^* + H^+ + e^- \rightarrow O^* + H_2O^*, \tag{8}$$

$$HOOH^* + H^+ + e^- \rightarrow HO^* + H_2O^*, \tag{9}$$

$$O^* + H_2O^* + H^+ + e^- \rightarrow HO^* + H_2O^*, \tag{10}$$

$$HO^* + H_2O^* + H^+ + e^- \rightarrow * + 2H_2O. \tag{11}$$

For the dissociative mechanism, the adsorbed $O_2^*$ can dissociate into two oxygen atoms before the protons and electrons transfer on the surface of substrates. The transition state

**Table 2** Bader point charge ($e^-$): positive and negative signs indicate the charge gain and loss, respectively. The charge difference ($\delta C$) (in $e^-$) after *versus* before the adsorption of the intermediates is presented in the parenthesis. Error for point charge calculation is of about 0.005 $e^-$

| System | Element[a] | FePor | FeTMP | FeTAP | FeTCP |
|---|---|---|---|---|---|
| Clean substrate | Fe | −1.369 | −1.359 | −1.394 | −1.391 |
| | $\sum$C | −8.952 | −8.664 | −12.717 | −9.513 |
| | $\sum$N | 10.345 | 10.323 | 10.295 | 10.567 |
| | $\sum$core | 0.024 | 0.300 | −3.816 | −0.337 |
| | $\sum$func | −0.024 | −0.300 | 3.816 | 0.337 |
| $O_2^*$/substrate | $O_2^*$ | 0.439 | 0.466 | 0.504 | 0.393 |
| | Fe | −1.533 (−0.164) | −1.542 (−0.183) | −1.511 (−0.117) | −1.523 (−0.132) |
| | $\sum$C | −9.093 (−0.141) | −8.756 (−0.092) | −13.153 (−0.436) | −9.413 (0.100) |
| | $\sum$N | 10.182 (−0.163) | 10.168 (−0.155) | 10.250 (−0.045) | 10.419 (−0.148) |
| | $\sum$core | −0.444 (−0.468) | −0.130 (−0.430) | −4.414 (−0.598) | −0.517 (−0.180) |
| | $\sum$func | 0.005 (0.029) | −0.336 (−0.036) | 3.910 (0.094) | 0.124 (−0.213) |
| HOO*/substrate | HOO* | 0.469 | 0.497 | 0.558 | 0.472 |
| | Fe | −1.552 (−0.183) | −1.563 (−0.204) | −1.540 (−0.146) | −1.595 (−0.204) |
| | $\sum$C | −9.139 (−0.187) | −8.920 (−0.256) | −13.055 (−0.338) | −9.562 (−0.049) |
| | $\sum$N | 10.229 (−0.116) | 10.200 (−0.123) | 10.277 (−0.018) | 10.421 (−0.146) |
| | $\sum$core | −0.462 (−0.486) | −0.283 (−0.583) | −4.318 (−0.502) | −0.736 (−0.399) |
| | $\sum$func | −0.007 (0.017) | −0.214 (0.086) | 3.760 (−0.056) | 0.264 (−0.073) |
| HOOH*/substrate | HOOH* | 0.517 | 0.667 | 0.691 | 0.613 |
| | Fe | −1.539 (−0.170) | −1.623 (−0.264) | −1.582 (−0.188) | −1.616 (−0.225) |
| | $\sum$C | −9.172 (−0.220) | −8.848 (−0.184) | −13.205 (−0.488) | −9.378 (0.135) |
| | $\sum$N | 10.185 (−0.160) | 10.249 (−0.074) | 10.268 (−0.027) | 10.234 (−0.333) |
| | $\sum$core | −0.526 (−0.550) | −0.222 (−0.522) | −4.519 (−0.703) | −0.760 (−0.423) |
| | $\sum$func | 0.009 (0.033) | −0.445 (−0.145) | 3.828 (0.012) | 0.147 (−0.190) |
| $O^* + H_2O^*$/substrate | $O^* + H_2O^*$ | 0.735 | 0.734 | 0.74 | 0.722 |
| | Fe | −1.642 (−0.273) | −1.625 (−0.266) | −1.565 (−0.171) | −1.640 (−0.249) |
| | $\sum$C | −9.147 (−0.195) | −8.74 (−0.076) | −13.133 (−0.416) | −9.378 (0.135) |
| | $\sum$N | 10.120 (−0.225) | 10.085 (−0.238) | 10.132 (−0.163) | 10.085 (−0.482) |
| | $\sum$core | −0.669 (−0.693) | −0.280 (−0.580) | −4.566 (−0.750) | −0.933 (−0.596) |
| | $\sum$func | −0.066 (−0.042) | −0.454 (−0.154) | 3.826 (0.010) | 0.211 (−0.126) |
| $HO^* + H_2O^*$/substrate | $HO^* + H_2O^*$ | 0.505 | 0.532 | 0.564 | 0.501 |
| | Fe | −1.572 (−0.203) | −1.589 (−0.230) | −1.522 (−0.128) | −1.588 (−0.197) |
| | $\sum$C | −9.013 (−0.061) | −8.709 (−0.045) | −13.088 (−0.371) | −9.280 (0.233) |
| | $\sum$N | 10.164 (−0.181) | 10.150 (−0.173) | 10.210 (−0.085) | 10.204 (−0.363) |
| | $\sum$core | −0.421 (−0.445) | −0.148 (−0.448) | −4.400 (−0.584) | −0.664 (−0.327) |
| | $\sum$func | −0.084 (−0.060) | −0.384 (−0.084) | 3.836 (0.020) | 0.163 (−0.174) |
| $2H_2O^*$/substrate | $2H_2O^*$ | −0.013 | −0.008 | −0.002 | −0.035 |
| | Fe | −1.422 (−0.053) | −1.424 (−0.065) | −1.399 (−0.005) | −1.420 (−0.029) |
| | $\sum$C | −9.006 (−0.054) | −8.663 (0.001) | −12.950 (−0.233) | −9.277 (0.236) |
| | $\sum$N | 10.411 (0.066) | 10.353 (0.030) | 10.351 (0.056) | 10.399 (−0.168) |
| | $\sum$core | −0.017 (−0.041) | 0.266 (−0.034) | −3.998 (−0.182) | −0.298 (0.039) |
| | $\sum$func | 0.030 (0.054) | −0.258 (0.042) | 4.000 (0.184) | 0.333 (−0.004) |
| $O^* + O^*$/substrate | $O^* + O^*$ | 1.599 | 1.506 | 1.56 | 1.546 |
| | Fe | −1.694 (−0.325) | −1.597 (−0.238) | −1.648 (−0.254) | −1.684 (−0.293) |
| | $\sum$C | −9.973 (−1.021) | −9.630 (−0.966) | −13.874 (−1.157) | −10.342 (−0.829) |
| | $\sum$N | 10.131 (−0.214) | 10.103 (−0.220) | 10.178 (−0.117) | 10.272 (−0.295) |
| | $\sum$core | −1.536 (−1.560) | −1.124 (−1.424) | −5.344 (−1.528) | −1.754 (−1.417) |
| | $\sum$func | −0.063 (−0.039) | −0.382 (−0.082) | 3.784 (−0.032) | 0.208 (−0.129) |
| $O^* + HO^*$/substrate | $O^* + HO^*$ | 1.312 | 1.305 | 1.318 | 1.276 |
| | Fe | −1.578 (−0.209) | −1.545 (−0.186) | −1.514 (−0.120) | −1.581 (−0.190) |
| | $\sum$C | −9.989 (−1.037) | −9.631 (−0.967) | −13.695 (−0.978) | −10.254 (−0.741) |
| | $\sum$N | 10.221 (−0.124) | 10.126 (−0.197) | 10.175 (−0.120) | 10.280 (−0.287) |
| | $\sum$core | −1.346 (−1.370) | −1.050 (−1.350) | −5.034 (−1.218) | −1.555 (−1.218) |
| | $\sum$func | 0.034 (0.058) | −0.255 (0.045) | 3.716 (−0.100) | 0.279 (−0.058) |
| $HO^* + HO^*$/substrate | $HO^* + HO^*$ | 1.144 | 1.120 | 1.139 | 1.116 |
| | Fe | −1.600 (−0.231) | −1.598 (−0.239) | −1.550 (−0.156) | −1.568 (−0.177) |
| | $\sum$C | −9.505 (0.553) | −9.154 (−0.490) | −13.510 (−0.793) | −9.868 (−0.355) |
| | $\sum$N | 10.084 (−0.261) | 9.978 (−0.345) | 10.144 (−0.151) | 9.993 (−0.574) |
| | $\sum$core | −1.021 (−1.045) | −0.774 (−1.074) | −4.916 (−1.100) | −1.443 (−1.106) |
| | $\sum$func | −0.123 (−0.099) | −0.346 (−0.046) | 3.777 (−0.039) | 0.327 (−0.010) |

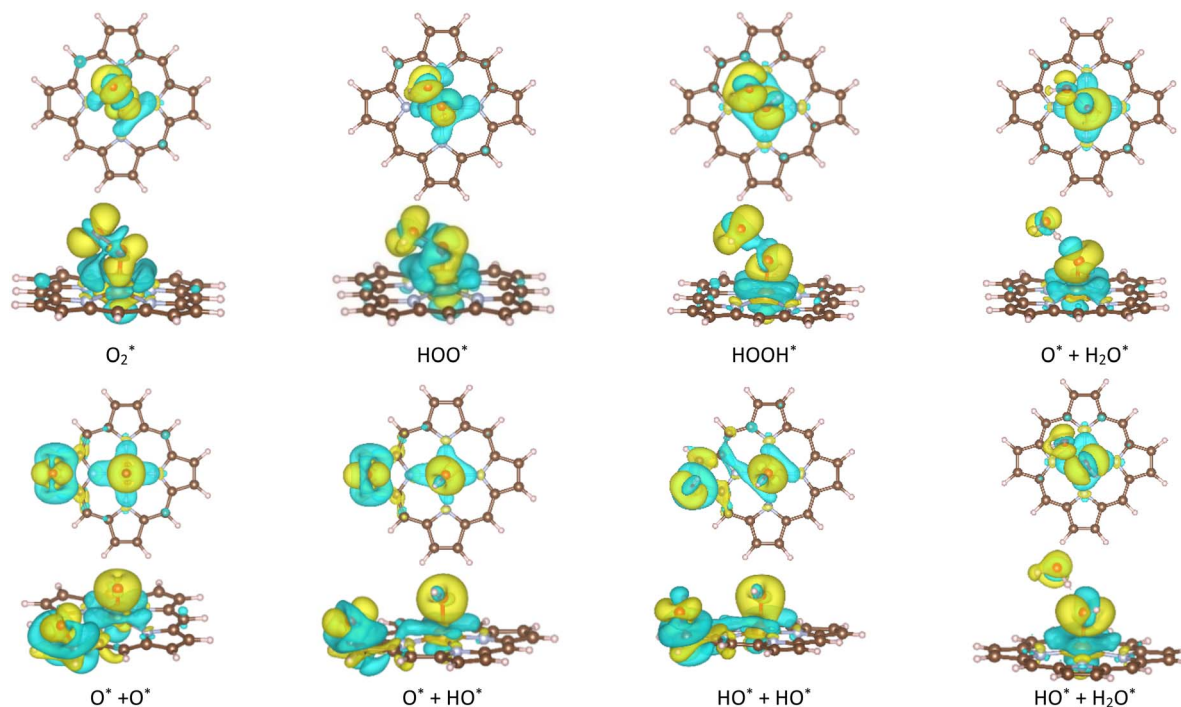[a] The $\sum$ denotation implies the summation of the charge.

**Fig. 3** The charge density difference of the [FePor + intermediate] systems. Accumulation charge (yellow) and donation charge (cyan). The other substrates also exhibit the same feature of charge accumulation for the oxygen atoms. The iso-surface value is 0.0015 e⁻Bohr$^{-3}$.
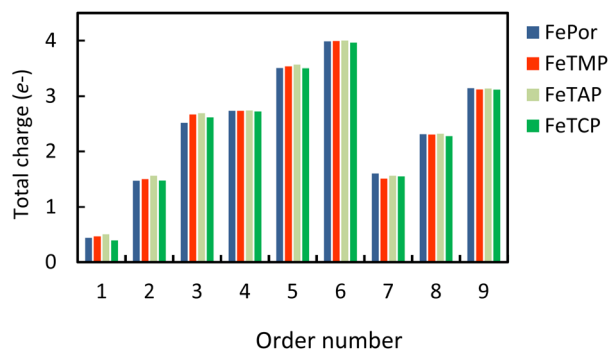


**Fig. 4** The total accumulating charge of two oxygen atoms of each intermediate. Blocks 1–9 presents $O_2^*$, HOO*, HOOH*, O* + $H_2O$*, HO* + $H_2O$*, $2H_2O$*, O* + O*, O* + HO*, and HO* + HO*, respectively.

$$O_2^* \rightarrow O^* + O^*, \quad (12)$$

$$O^* + O^* + H^+ + e^- \rightarrow O^* + HO^*, \quad (13)$$

$$O^* + HO^* + H^+ + e^- \rightarrow HO^* + HO^*, \quad (14)$$

$$O^* + HO^* + H^+ + e^- \rightarrow O^* + H_2O^*, \quad (15)$$

$$HO^* + HO^* + H^+ + e^- \rightarrow HO^* + H_2O^*, \quad (16)$$

$$O^* + H_2O^* + H^+ + e^- \rightarrow HO^* + H_2O^*, \quad (17)$$

$$HO^* + H_2O^* + H^+ + e^- \rightarrow {}^* + 2H_2O. \quad (18)$$

(TS) structure of the $O_2$ dissociation was determined *via* three steps: first, the initial state (IS) of $O_2^*$ and the final state (FS) of O* + O* were found by the DFT calculations, then the TS structure was interpolated based on the initial and final state structures. After that, the TS structure was optimized using the NEB – DFT method, and the total energy of the optimized TS structure was calculated. As shown in Fig. 7, the $O_2^*$ dissociation must overcome an activation barrier of 4.32, 2.67, 1.67, and 2.73 eV on FePor (a), FeTMP (b), FeTAP (c), and FeTCP (d), respectively. It should be emphasized that the barrier obtained by NEB only includes the electronic effect (disregard zero-point energy, entropy, and electrode potential). The dissociative mechanism can proceed in the following reaction series:[29,33]
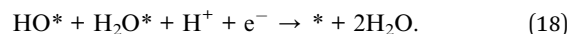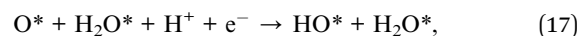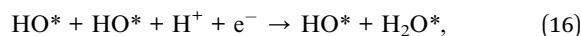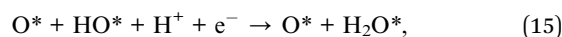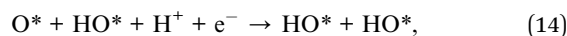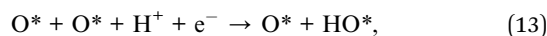
The Gibbs free energy for each intermediate step, relative to that of the isolated by-product $2H_2O$ in eqn (11) and (18), was calculated at the temperature of 300 K, the pressure of 1 bar, pH = 0, and the electrode potential of $U$ = 1.23 V. The zero-point energy and entropy changes ($\Delta ZPE$ and $T\Delta S$) of each intermediate reaction were calculated by subtracting the corresponding values of the reactants from that of the products in the reversible hydrogen electrode conditions ($H^+ + e = \frac{1}{2}H_2$).

Table 3 presents the obtained values for the Gibbs free energy of the ORR intermediate steps in the associative and dissociative pathways, see eqn (5)–(18). The Gibbs free energy for the transition state (TS) of the $O_2^*$ dissociation into O* + O* was calculated in the same manner as those for the ORR intermediate steps by including the contributions of zero-point energy, entropy, and electrode potential. Fig. 8 and 9 show the
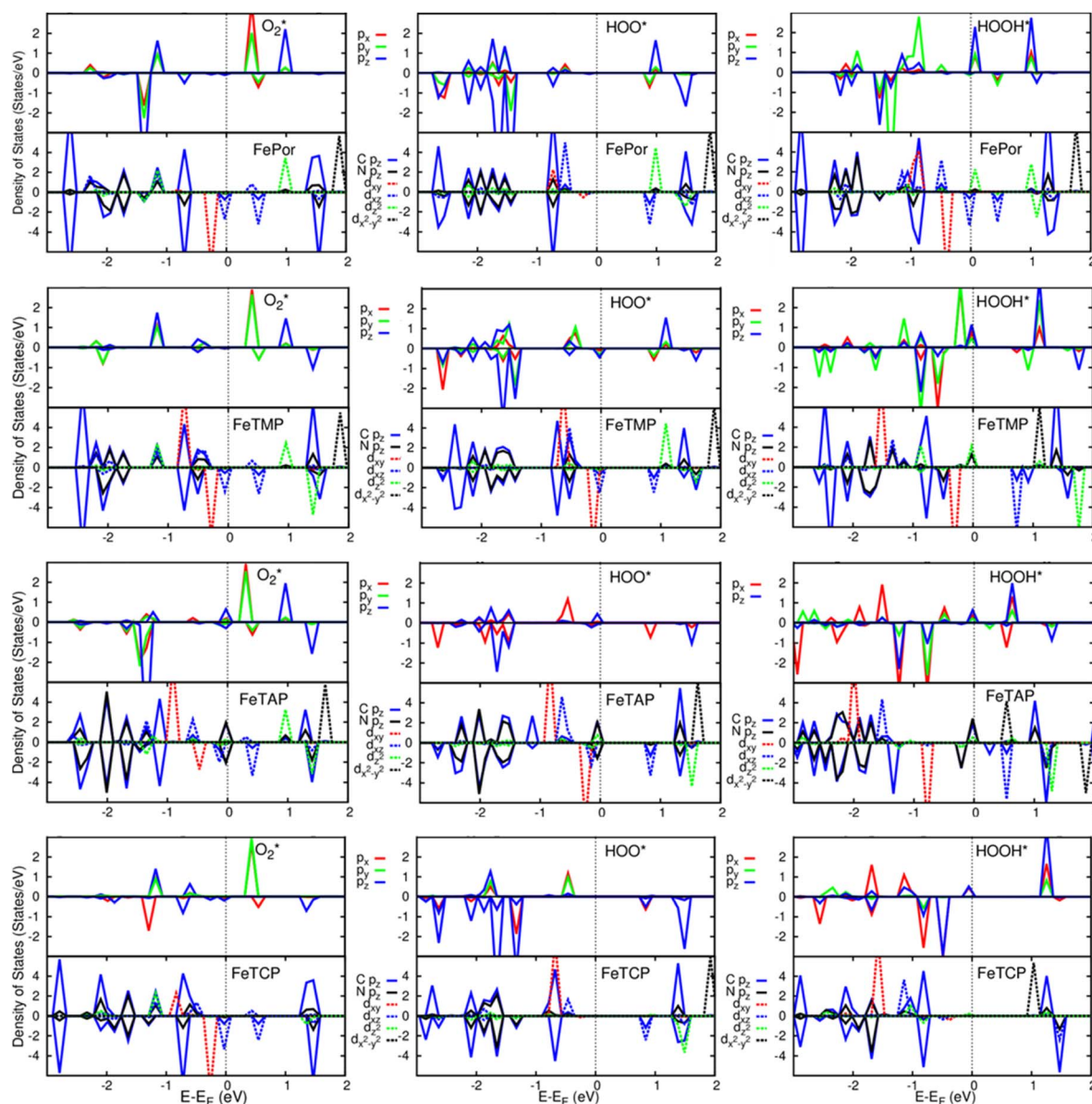
**Fig. 5** Electronic density of states for the systems of the substrates with $O_2^*$, HOO*, and HOOH*. The $d_{xz}$ and $d_{yz}$ orbitals are identical.



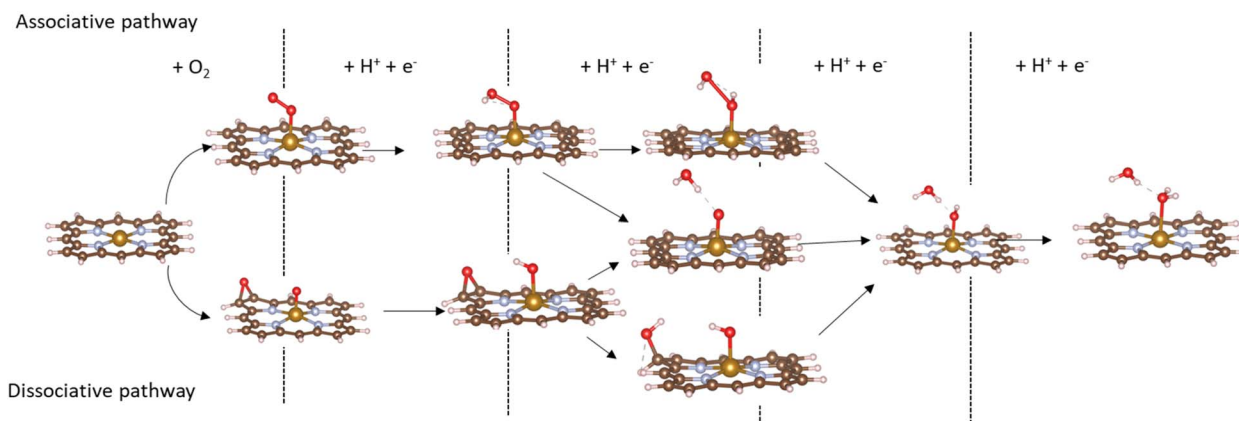**Fig. 6** The dissociative (above) and associative (below) pathways.

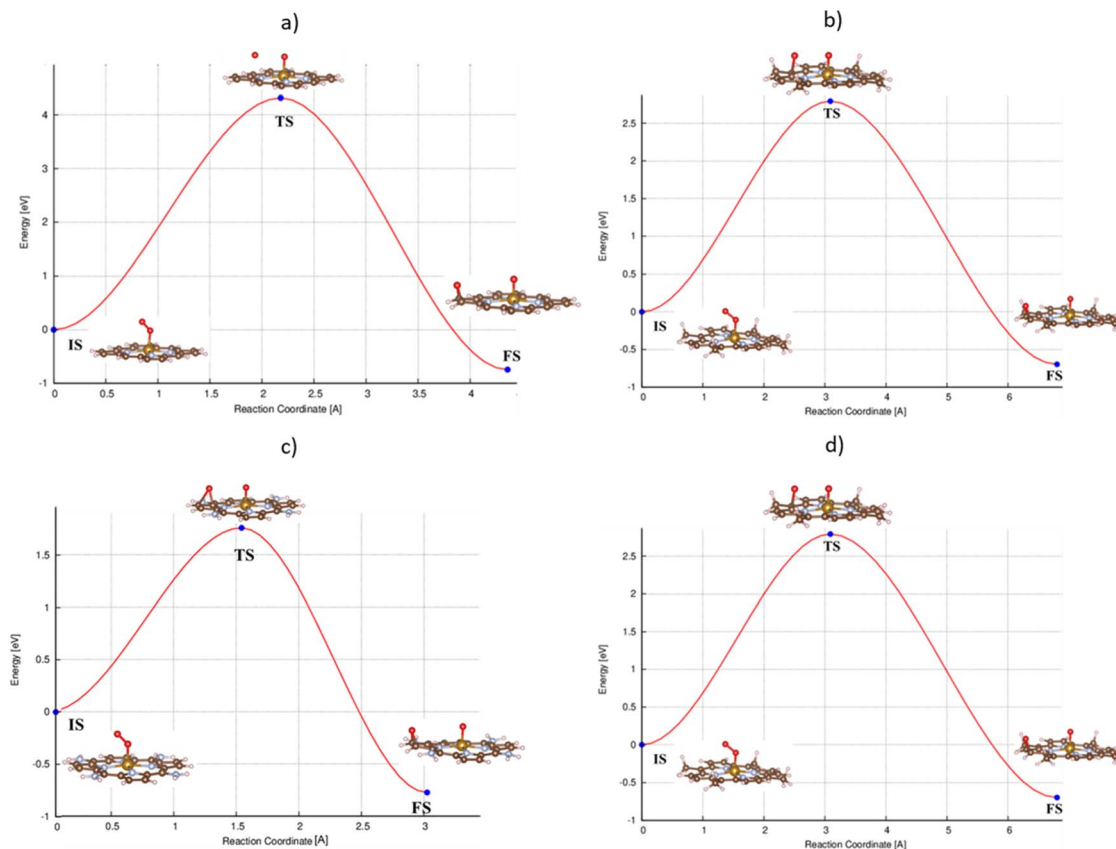© 2023 The Author(s). Published by the Royal Society of Chemistry

**Fig. 7** The minimum energy path from the initial state (IS) to the final state (FS) of the $O_2^*$ dissociation must overcome an activation barrier of 4.32, 2.67, 1.67, and 2.73 eV at the transition state (TS) on FePor (a), FeTMP (b), FeTAP (c), and FeTCP (d), respectively.

**Table 3** Gibbs free energies (eV), relative to that of isolated $2H_2O$, for different intermediates in the associative and dissociative pathways at the equilibrium electrode potential of 1.23 V, and $H^+ + e^- = \frac{1}{2}H_2$

| ORR intermediates | FePor | FeTMP | FeTAP | FeTCP |
|---|---|---|---|---|
| $O_2^* + 2H_2$ | −1.09 | −1.05 | −1.31 | −0.99 |
| $HOO^* + 3/2H_2$ | −0.24 | −0.20 | −0.52 | −0.44 |
| $HOOH^* + H_2$ | 0.95 | 0.78 | 0.55 | 0.85 |
| $O^* + H_2O^* + H_2$ | −0.87 | −0.87 | −1.27 | −0.79 |
| $HO^* + H_2O^* + \frac{1}{2}H_2$ | −0.09 | −0.07 | −0.46 | −0.02 |
| $2H_2O$ | 0 | 0 | 0 | 0.00 |
| $TS + 2H_2$ | 1.99 | 1.14 | 0.39 | 1.40 |
| $O^* + O^* + 2H_2$ | −1.65 | −1.58 | −1.78 | −1.54 |
| $O^* + HO^* + 3/2H_2$ | −0.95 | −0.84 | −1.11 | −0.83 |
| $HO^* + HO^* + H_2$ | 0.13 | 0.08 | −0.41 | 0.14 |

free energy diagrams for the associative and dissociative pathways, respectively. We found that the ORR on all substrates is going downhill from $O_2 + 2H_2$ to $O_2^* + 2H_2$, which implies that the $+ 2H_2$ step can proceed automatically without an energy supply for both reaction pathways.

Fig. 8 shows that, in the forwarding direction from left to right, the first hydrogenation step from $O_2^* + 2H_2$ to $HOO^* + 3/2H_2$, the second hydrogenation from $HOO^* + 3/2H_2$ to $HOOH^* + H_2$, the third hydrogenation from $O^* + H_2O^* + H_2$ to $HO^* + H_2O^* + \frac{1}{2}H_2$, and the fourth hydrogenation from $HO^* + H_2O^* +$

$\frac{1}{2}H_2$ to $2H_2O$ are going uphill. This result implies that generating the corresponding intermediates, i.e., HOO*, HOOH*, HO* + $H_2O^*$, and $2H_2O^*$ requires activation energy. Besides, the other hydrogenation steps, from $HOO^* + 3/2H_2$ to $O^* + H_2O^* + H_2$, and $HOOH^* + H_2$ to $HO^* + H_2O^* + \frac{1}{2}H_2$, can proceed automatically because they are going downhill. Furthermore, the formation of $O^* + H_2O^* + H_2$ occurs more easily than that of HOOH*, even though they contain the same number of
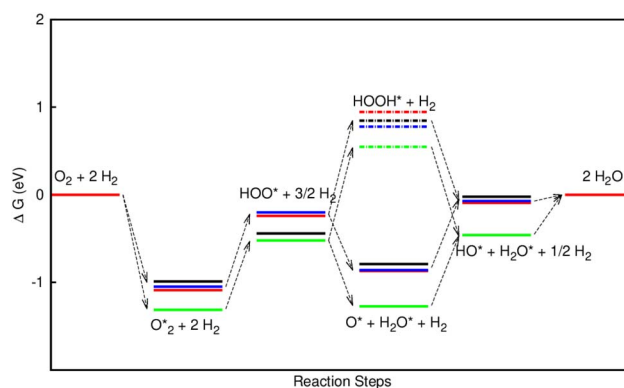


**Fig. 8** Gibbs free energy diagram for the oxygen reduction reaction in the associative mechanism at the electrode potential of $U = 1.23$ V on FePor (red), FeTMP (blue), FeTAP (green), and FeTCP (black).

**Table 4** The thermodynamic barrier (eV) for the uphill steps of the ORR at $U = 1.23$ V, which is the Gibbs free energy difference between the uphill step and the previous one, and $H^+ + e^- = \frac{1}{2}H_2$

| ORR intermediates | FePor | FeTMP | FeTAP | FeTCP |
|---|---|---|---|---|
| $O_2^* + 2H_2 \rightarrow HOO^* + 3/2H_2$ | 0.84 | 0.85 | 0.79 | 0.55 |
| $HOO^* + 3/2H_2 \rightarrow HOOH^* + H_2$ | 1.19 | 0.98 | 1.07 | 1.29 |
| $O^* + H_2O^* + H_2 \rightarrow HO^* + H_2O^* + \frac{1}{2}H_2$ | 0.78 | 0.80 | 0.81 | 0.77 |
| $HO^* + H_2O^* + \frac{1}{2}H_2 \rightarrow 2H_2O$ | 0.09 | 0.07 | 0.46 | 0.02 |
| $O_2^* + 2H_2 \rightarrow O^* + O^* + 2H_2$ | 3.08 | 2.18 | 1.7 | 2.39 |
| $O^* + O^* + 2H_2 \rightarrow O^* + HO^* + 3/2H_2$ | 0.70 | 0.74 | 0.67 | 0.71 |
| $O^* + HO^* + 3/2H_2 \rightarrow HO^* + HO^* + H_2$ | 1.08 | 0.92 | 0.70 | 0.97 |
| $O^* + HO^* + 3/2H_2 \rightarrow O^* + H_2O^* + H_2$ | 0.08 | | | 0.04 |

hydrogen atoms, and both can convert into HO* + H$_2$O*. The conversion of HO* + H$_2$O* into 2H$_2$O requires a smaller activation energy on FePor, FeTMP, and FeTCP than that on FeTAP. The thermodynamic barriers for the different reaction steps were listed in Table 4. From the left to right side of the Gibbs free energy diagram in Fig. 8, the highest thermodynamic barrier was found at the second hydrogenation step that converts HOO* to HOOH*. Therefore, the rate-limiting step in the associative mechanism should be the second hydrogenation step. The finding of HOOH* on the substrates supports the experimental finding,[21] where both H$_2$O$_2$ and H$_2$O were the products of the reaction.

In the dissociative mechanism, we incorporated the transition state of the O$_2^*$ dissociation to 2O* into the Gibbs free energy diagram, Fig. 9. This figure demonstrates that the formation of O* + O* + 2H$_2$ from O$_2^*$ + 2H$_2$ must overcome a large energy barrier. The first hydrogenation step from O* + O* + 2H$_2$ to O* + HO* + 3/2H$_2$, the second hydrogenation from O* + HO* + 3/2H$_2$ to HO* + HO* + H$_2$, and the third hydrogenation step from O* + H$_2$O* + H$_2$ to HO* + H$_2$O* + $\frac{1}{2}$H$_2$, are going uphill and request the corresponding activation barrier (Table 4). However, these thermodynamic barriers for the hydrogenation steps are significantly lower than that of the O$_2^*$ dissociation. Therefore, the rate-limiting step in the dissociative mechanism locates at the dissociation step of the O$_2^*$ intermediate into 2O*. In the backward direction from the right to the

left side of the diagrams in Fig. 8 and 9, the conversion from 2H$_2$O to HO* + H$_2$O* + $\frac{1}{2}$H$_2$ to O* + H$_2$O* + H$_2$ occurs automatically on all substrates because the O* + H$_2$O* and HO* + H$_2$O* intermediates adsorbed too strongly as shown in Table 1. This result agrees with the Sabatier principle, which states that the adsorption strength between a catalyst and reactant should be neither too strong nor too weak for good catalytic performance. Therefore, there are possibilities that these intermediates will be long-lasting on the surface and block the catalytic active sites of the substrates. This result is also in good agreement with the theoretical finding that the HO* intermediate always occurs on the porphyrin.[32] Based on the thermodynamic energy barriers, we found that the associative mechanism on FeTMP requires the lowest activation energy. Similarly, the dissociative mechanism needs to overcome the smallest activation barrier on FeTAP. Even though, the reconversion of 2H$_2$O into the more stable intermediates perhaps hinders the catalytic performance of all substrates for both associative and dissociative mechanisms. Combining with the second lowest activation barrier for both mechanisms, we can conclude that FeTMP and FeTAP should be the best choices among the considered substrates for the oxygen reduction reaction.

Furthermore, the highest electrode potential at which the molecular oxygen spontaneously reduced to 2H$_2$O is the onset potential.[30] Our calculations obtained the onset potential of 0.39, 0.38, 0.42, and 0.46 V for the associative mechanism, and 0.53, 0.49, 0.56, and 0.52 V for the dissociative mechanism on FePor, FeTMP, FeTAP, and FeTCP, respectively. The results implied that the onset potential for the ORR varies with the functional groups. Particularly, 0.39 and 0.53 V for the associative and dissociative pathways on FePor agree with 0.67 V obtained by the previous experiment.[21]

## 4. Conclusions

The present work studied the oxygen reduction reaction on the original porphyrin and the porphyrin with different functional groups using density functional theory calculations and a thermodynamic model. The study has been performed in the following steps: (1) searched for the possible intermediates and their favorable adsorption configurations. (2) Clarified the interaction between the intermediates with the porphyrin substrates. (3) Proposed the reaction mechanisms, the
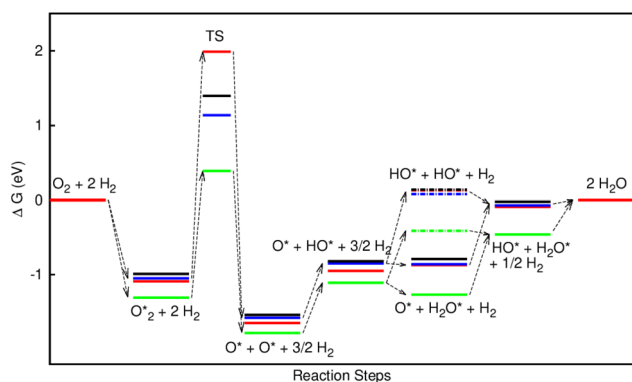


**Fig. 9** Gibbs free energy diagram for the oxygen reduction reaction following the dissociative mechanism at the electrode potential $U = 1.23$ V on FePor (red), FeTMP (blue), FeTAP (green), and FeTCP (black).

associative and dissociative ones. (4) Calculated the Gibbs free energy for each intermediate step in the proposed mechanisms. (5) Plotted the Gibbs free energy diagrams for each mechanism. The analysis showed that the rate-limiting step for the associative and dissociative mechanisms is the second hydrogenation step converting HOO to HOOH and the dissociation of $O_2$ into 2O, respectively. The $-CH_3$ and $-NH_2$ groups offered small activation barriers; therefore, better catalytic activity for oxygen reduction reaction among the considered functional groups and the original porphyrin.

## Author contributions

Conceptualization (DNS), formal analysis (TPD and DNS), investigation (TPD), resources (VC and DNS), supervision (VC and DNS), validation (DNS), visualization (TPD), writing manuscript (DNS), reviewing and editing (TPD, VC, and DNS).

## Conflicts of interest

There are no conflicts to declare.

## Acknowledgements

## References

1 I. Staffell, D. Scamman, A. Velazquez Abad, P. Balcombe, P. E. Dodds, P. Ekins, N. Shah and K. R. Ward, The role of hydrogen and fuel cells in the global energy system, *Energy Environ. Sci.*, 2019, **12**, 463–491.

2 X. Cheng, Z. Shi, N. Glass, L. Zhang, J. Zhang, D. Song, *et al*, A review of PEM hydrogen fuel cell contamination: Impacts, mechanisms, and mitigation, *J. Power Sources*, 2007, **165**, 739–756.

3 V. R. Stamenkovic, B. S. Mun, M. Arenz, *et al*, Trends in electrocatalysis on extended and nanoscale Pt-bimetallic alloy surfaces, *Nat. Mater.*, 2007, **6**, 241–247.

4 Y. Zhu, S. Wang, Q. Luo, H. Huang, S. Tang and Y. Du, Facile synthesis of structurally ordered low-Pt-loading Pd–Pt–Fe nanoalloys with enhanced electrocatalytic performance for oxygen reduction reaction, *J. Alloys Compd.*, 2021, **855**, 157322.

5 E. Daş, S. A. Gürsel and A. B. Yurtcan, Pt-alloy decorated graphene as an efficient electrocatalyst for PEM fuel cell reactions, *J. Supercrit. Fluids*, 2020, **165**, 104962.

6 H. Yang, W. Vogel, C. Lamy and N. Alonso-Vante, Structure and electrocatalytic activity of carbon-supported Pt-Ni alloy nanoparticles toward the oxygen reduction reaction, *J. Phys. Chem. B*, 2004, **108**, 11024–11034.

7 J. Liu, J. Lan, L. Yang, F. Wang and J. Yin, PtM (M = Fe, Co, Ni) bimetallic nanoclusters as active, methanol tolerant, and stable catalysts toward oxygen reduction reaction, *ACS Sustainable Chem. Eng.*, 2019, **7**, 6541–6549.

8 Y. Li, Q. Li, H. Wang, L. Zhang, D. P. Wilkinson and J. Zhang, Recent progresses in oxygen reduction reaction electrocatalysts for electrochemical energy applications, *Electrochem. Energy Rev.*, 2019, **2**, 518–538.

9 O. Savadogo, K. Lee, S. Mitsushima, N. Kamiya and K.-I. Ota, Investigation of some new palladium alloys catalysts for the oxygen reduction reaction in an acid medium, *J. New Mater. Electrochem. Syst.*, 2004, **7**, 77–84.

10 M. D. Bhatt, G. Lee and J. S. Lee, Screening of oxygen-reduction-reaction-efficient electrocatalysts based on Ag–M (M= 3d, 4d, and 5d Transition Metals) nanoalloys: a density functional theory study, *Energy Fuels*, 2017, **31**, 1874–1881.

11 T. Wang, A. Chutia, D. J. L. Brett, P. R. Shearing, G. He, G. Chai and I. P. Parkin, Palladium alloys used as electrocatalysts for the oxygen reduction reaction, *Energy Environ. Sci.*, 2021, **14**, 2639–2669.

12 D. N. Son, P. N. Thanh, N. D. Quang, K. Takahashi and M. P. Pham-Ho, *J. Appl. Electrochem.*, 2017, **47**, 747–754.

13 R. L. Borup, *et al*, Recent developments in catalyst-related PEM fuel cell durability, *Curr. Opin. Electrochem.*, 2020, **21**, 192–200.

14 H.-J. Niu, L. Zhang, J.-J. Feng, Q.-L. Zhang, H. Huang and A.-J. Wang, Graphene-encapsulated cobalt nanoparticles embedded in porous nitrogen-doped graphitic carbon nanosheets as efficient electrocatalysts for oxygen reduction reaction, *J. Colloid Interface Sci.*, 2019, **552**, 744–751.

15 Y. He, S. Liu, C. Priest, Q. Shi and G. Wu, Atomically dispersed metal–nitrogen–carbon catalysts for fuel cells: advances in catalyst design, electrode performance, and durability improvement, *Chem. Soc. Rev.*, 2020, **49**, 3484–3524.

16 S. Hiroto, Y. Miyake and H. Shinokubo, Synthesis and functionalization of porphyrins through organometallic methodologies, *Chem. Rev.*, 2017, **117**, 2910–3043.

17 C. T. Carver, B. D. Matson and J. M. Mayer, Electrocatalytic oxygen reduction by iron tetra-arylporphyrins bearing pendant proton relays, *J. Am. Chem. Soc.*, 2012, **134**, 5444–5447.

18 B. D. Matson, C. T. Carver, A. Von Ruden, J. Y. Yang, S. Raugei and J. M. Mayer, Distant protonated pyridine groups in water-soluble iron porphyrin electrocatalysts promote selective oxygen reduction to water, *Chem. Commun.*, 2012, **48**, 11100–11102.

19 Q. He, *et al*, Molecular catalysis of the oxygen reduction reaction by iron porphyrin catalysts tethered into Nafion layers: an electrochemical study in solution and a membrane-electrode-assembly study in fuel cells, *J. Power Sources*, 2012, **216**, 67–75.

20 C. Shi, B. Steiger, M. Yuasa and F. C. Anson, Electroreduction of $O_2$ to $H_2O$ at unusually positive potentials catalyzed by the simplest of the cobalt porphyrins, *Inorg. Chem.*, 1997, **36**, 4294–4295.

21 G. Tei, T. Tamaki, T. Hayashi, K. Nakajima, A. Sakai, S. Yotsuhashi and T. Ogawa, Oxygen reduction reaction (ORR) activity of a phenol-substituted linear FeIII–porphyrin dimer, *Eur. J. Inorg. Chem.*, 2017, **26**, 3229–3232.

22 M. Yuasa, R. Nishihara, C. Shi and F. C. Anson, A comparison of several meso-tetraalkyl cobalt porphyrins as catalysts for the electroreduction of dioxygen, *Polym. Adv. Technol.*, 2001, **12**, 266–270.

23 F. A. Walker, D. Beroiz and K. M. Kadish, Electronic effects in transition metal porphyrins. 2. The sensitivity of redox and ligand addition reactions in para-substituted tetraphenylporphyrin complexes of cobalt(II), *J. Am. Chem. Soc.*, 1976, **98**, 3484–3489.

24 Sk Amanullah, S. A. H. A. Paramita and D. E. Y. Abhishek, $O_2$ reduction by iron porphyrins with electron withdrawing groups: to scale or not to scale, *Faraday Discuss.*, 2022, **234**, 143–158.

25 J. Hafner, Ab-initio simulations of materials using VASP: density-functional theory and beyond, *J. Comput. Chem.*, 2008, **29**, 2044–2078.

26 P. E. Blöchl, Projector augmented-wave method, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 1994, **50**, 17953–17979.

27 H. J. Monkhorst and J. D. Pack, Special points for Brillouin-zone integrations, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 1976, **13**, 5188–5192.

28 O. My Na, N. T. Xuan Huynh, P. T. Thi, V. Chihaia and D. N. Son, Mechanism and activity of the oxygen reduction reaction on $WTe_2$ transition metal dichalcogenide with Te vacancy, *RSC Adv.*, 2020, **10**, 8460–8469.

29 D. N. Son, O. K. Le, V. Chihaia and K. Takahashi, Effects of Co content in Pd-Skin/PdCo alloys for oxygen reduction reaction: density functional theory predictions, *J. Phys. Chem. C*, 2015, **119**, 24364–24372.

30 J. K. Nørskov, J. Rossmeisl, A. Logadottir, L. Lindqvist, J. R. Kitchin, T. Bligaard and H. Jonsson, Origin of the overpotential for oxygen reduction at a fuel-cell cathode, *J. Phys. Chem. B*, 2004, **108**, 17886–17892.

31 J. P. Collman, J. I. Brauman and K. S. Suslick, Oxygen binding to iron porphyrins, *J. Am. Chem. Soc.*, 1975, **97**(24), 7185–7186.

32 Y. Wang, Y. Tang and K. Zhou, Self-adjusting activity induced by intrinsic reaction intermediate in Fe−N−C single-atom catalysts, *J. Am. Chem. Soc.*, 2019, **141**, 14115–14119.

33 D. N. Son and K. Takahashi, Selectivity of palladium–cobalt surface alloy toward oxygen reduction reaction, *J. Phys. Chem. C*, 2012, **116**, 6200–6207.