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Tailor-designed binary Ni–Cu nano dendrites decorated 3D-carbon felts for efficient glycerol electrooxidation

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Herein, 3D-Carbon Felt (CF) are decorated with nickel–copper (Ni–Cu@CF) bimetallic nanostructures through either sequential or co-electrodeposition tactics. Their catalytic activity towards glycerol electrooxidation is investigated by employing cyclic voltammetry (CV) and linear sweep voltammetry LSV. The morphology and composition of the various Ni–Cu@CF are investigated using X-ray diffraction (XRD) and field emission scanning electron microscopy (FE-SEM) together with various electrochemical measurements (e.g., CV, chronoamperometry, LSV). The co-deposition of Ni–Cu shows a dendritic-like structure with higher electrocatalytic activity towards glycerol electrooxidation compared to the monometallic counterparts. Interestingly, the best electrode (NiCu@CF Ni particles as the top layer) prepared by sequential electrodeposition shows 1.6-fold higher glycerol oxidation activity, manifested in oxidation current, compared to Ni-coated CF due to Ni particles covering the surface of dendritic copper uniformly. Thus, the surface concentration of Ni is increased and at the same time a synergistic effect occurs between Ni and Cu by the simple addition of Cu which reinforces the surface concentration of Ni from 3.4×10^{-8} to 1.1×10^{-7} mol cm⁻².

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1. Introduction

Direct alcohol fuel cells (e.g., methanol fuel cells) have advantages over traditional hydrogen fuel cells including better safety, high energy density, and easy handling and storing. Despite the aforementioned advantages of methanol fuel cells, they still suffer some drawbacks including high toxicity, high volatility, and poor stability of their commercial Pt-based electrodes, which is attributed to their rapid poisoning with *in situ* generated CO-like intermediates.^{1–8}

Glycerol seems to be a promising fuel candidate for liquid fuel cells, due to its unique features including non-volatility, lower toxicity, high boiling point, high theoretical energy density (6.4 kW h L⁻¹), and low cost since it is a by-product of the conversion of triglyceride into biodiesel (2 million ton introduced into the market per year).^{9–13} Additionally, the glycerol can be oxidized into highly valuable products such as dihydroxyacetone, tartaric acid, mesoxalate, glycolic acid, and dihydroxyacetone, which can be used in many industrial fields including drug delivery, complexing agents for heavy metals, polymer synthesis, and medical applications.^{5,10,14–17}

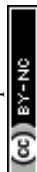
Nonprecious metal-based materials (e.g., Co, Fe, Cu, Ni, and their alloys) have been considered as promising cheap alternative

fuel cell electrodes for the costly, poor performance noble metal-based electrodes (i.e., Pt and Pd)^{18–21} Among them, Ni-based catalysts are the most promising non-platinum electrocatalysts attributing to their satisfied catalytic efficiency (with respect to activity and stability) towards the oxidation of various organic compounds including methanol,^{22–25} ethanol,^{26–28} glycerol^{17,21,29–37} urea,^{38,39} glucose,⁴⁰ etc. Many researchers have reported on improving the electrocatalytic performance of Ni electrocatalysts in direct glycerol fuel cells (DGFCs) by alloying it with second metal including Cu, Fe, Co, etc.^{20,21,41,42} For instance, El-Nagar *et al.*⁴³ investigated the glycerol electrooxidation at Ni-doped electrodeposited copper foams *via* the dynamic hydrogen bubbles technique. B. Habibi *et al.*⁴¹ studied electrooxidation of glycerol at Cu–Co alloys. Both have observed that the addition of Co or Cu to Ni improved its electrocatalytic activity and stability toward glycerol electrooxidation. Oliveira *et al.*²⁰ showed that Ni bimetallic and trimetallic surfaces (e.g., NiCo, NiFe and NiCoFe) exhibited better activity and stability for glycerol electrocatalysis attributing to the formation of metal oxides and hydroxides.

In this study, Cu is selected to modify Ni attributing to similar lattice parameters ($a = 3.616$ for Cu and 3.523 for Ni) allowing a wide range of Ni–Cu compositions with superior physical and chemical properties together with better electrocatalytic performance.^{41,44,45} Cu has a lower glycerol electrooxidation onset potential than Ni and can selectively electrooxidize glycerol towards a highly valuable product (dihydroxyacetone).⁴⁶ However, Cu suffers low stability and high charge

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transfer resistance as compared to Ni. Thus, herein, Cu is added to Ni to gain the advantages of both metals and minimize their disadvantages.

3D Carbon Felt (CF) is used as a substrate for the electrodeposited NiCu nanostructures attributing to its good stability, high specific surface area, low cost, corrosion resistance, high conductivity, and large potential window.⁴⁷ Besides, its electrochemical activity can be enhanced by several simple and fast methods (acid treatment, electrochemical treatment, plasma treatment, and thermal treatment).^{4,48} Mohamed E. Ghaith *et al.*⁴ investigated the impact of electrooxidation of CF either in 1 M sulfuric acid or 1 M sodium hydroxide at different potentials on the performance of the electrodeposition of Ni particles towards glycerol electrooxidation. The pretreatments in either acidic or alkaline improved the catalytic activity of the electrodeposited Ni nanoparticles by 2.5 times compared to non-pretreated decorated carbon felt.

Herein, commercial-available 3D-carbon felt has been decorated with Ni-Cu bimetallic nanostructures *via* two different electrodeposition strategies, *i.e.*, co-electrodeposition and sequential electrodeposition of Cu and Ni. Various electrodeposition conditions including deposition bath composition, the amount and the order of the electro-deposited catalyst have been optimized with an aim to maximize its efficiency towards glycerol electro-oxidation in alkaline medium.

2. Experimental

2.1. Materials

All the used chemicals are of analytical grade and were used as received without further purification. All the solutions were prepared using second distilled water. Sodium hydroxide (NaOH), nickel sulfate hexahydrate ($\text{NiSO}_4 \cdot 6\text{H}_2\text{O}$, 99.999%), and CuSO_4 anhydrous (99.999%) were purchased from Sigma-Aldrich.

2.2. Electrode preparation

Before use of commercial carbon felt cleaned by ultrasonication in ethanol for 10 min then rinsed by water then activated as the previous published paper⁴ then decorated with bimetallic Ni-Cu particles *via* two different electrodeposition protocols as described below:

I Sequential electrodeposition approach. Ni nanoparticles were first electrodeposited from 0.1 M Na_2SO_4 containing 4 mM NiSO_4 at -0.6 V (*vs.* SCE) and then followed by Cu nanoparticles electrodeposition from 0.1 M Na_2SO_4 solution containing 4 mM CuSO_4 at -0.6 V (*vs.* SCE). This electrode is assigned as CuNi@CF electrode (Cu is the top layer), while NiCu@CF is referring to the decorated CF with Ni particles as the top layer and Cu underneath layer. For comparison, Cu particles decorated CF (Cu@CF) and Ni particles decorated CF (Ni@CF) were synthesized using the same above-provided deposition conditions.

II Co-electrodeposition approach. Ni and Cu were simultaneously electrodeposited from 0.1 M Na_2SO_4 solution containing $\text{CuSO}_4/\text{NiSO}_4$ bath with different Cu : Ni ratios and fixed total concentration (4 mM). The obtained electrodes have given

a general symbol $\text{Ni}_x\text{Cu}_y\text{@CF}$, where X and Y are the ratios of Ni and Cu in the electrodeposition bath, respectively.

Next, all the as-prepared electrodes were activated in 0.2 M NaOH for 40 cycles from 0.0 to 0.7 V with a potential scan rate of 200 mV s^{-1} followed by 2 cycles in the same electrolyte with a lower potential scan rate (10 mV s^{-1}). During the preparation of all electrodes, no hydrogen evolution is noticed which means the hydrogen evolution can be ignored. So, we assume the deposition efficiency is 100%.

2.3. Material and electrochemical characterizations

2.3.1 Electrochemical measurements. Bio-logic potentiostat (model VSP-300) was used to perform all electrochemical measurements where decorated CF strip with different bimetallic nanostructures (with $0.3 \text{ cm} \times 0.2 \text{ cm}$ dimensions) and connected by GC rod (1 mm diameter), and saturated calomel electrode (SCE) served as working and reference electrodes, respectively, where the graphite rod serves as a counter electrode. The electrochemical activity and stability of the prepared electrocatalyst materials were evaluated by using cyclic voltammetry (CV), linear sweep voltammetry (LSV), and chronoamperometry in 0.2 M NaOH aqueous solution containing 8 mM glycerol, respectively.

2.3.2 Material characterization. The crystallographic orientation, morphology, and chemical composition of the prepared catalysts were investigated, respectively, by X-ray diffraction technique (XRD, Cu K α radiation, STOE STADI), field emission scanning electron microscope (FE-SEM, QUANTA FEG 250) coupled with an energy dispersive X-ray spectrometer (EDX) unit and XPS (X-ray photoelectron spectroscopy), using a CLAM4 electron analyzer from Thermo VG scientific, and an Mg K α X-ray source (1253.6 eV) XR 50 from SPECS.

3. Results and discussion

3.1. Material characterization

The morphology and the bulk composition of the sequentially and co-electrodeposited Cu-Ni bimetallic nanostructures were first examined by SEM-EDX to investigate the effect of used electrodeposition protocols on the shape and composition of the obtained bimetallic surfaces.

Fig. 1(A-F) shows the SEM images of Ni@CF (Fig. 1A), Cu@CF (Fig. 1B), $\text{Ni}_{85\%}\text{Cu}_{15\%}\text{@CF}$ (Fig. 1C), NiCu@CF (Fig. 1D), $\text{Ni}_{50\%}\text{Cu}_{50\%}\text{@CF}$ (Fig. 1E), and $\text{Ni}_{25\%}\text{Cu}_{75\%}\text{@CF}$ (Fig. 1F). As seen in this figure, while Ni@CF exhibited irregular shape-like-structure un-homogenously distributed over carbon felts, Cu@CF showed a horizontal dendritic-like structure at some positions and quite better distribution over the entire CF. On the other hand, NiCu@CF (prepared by sequential deposition approach) shows homogeneously distributed Ni particles covering the distributed vertical dendrites which is evidenced from the spot and mapping EDX of NiCu@CF at (Fig. 1K(c) and (G-J)) in which the intensity of Ni and Cu are comparable to each other. While $\text{Ni}_{85\%}\text{Cu}_{15\%}\text{@CF}$ (synthesized by co-electrodeposition strategy) electrode shows bad Ni cover to the well-distributed dendrites which is evidenced from the spot



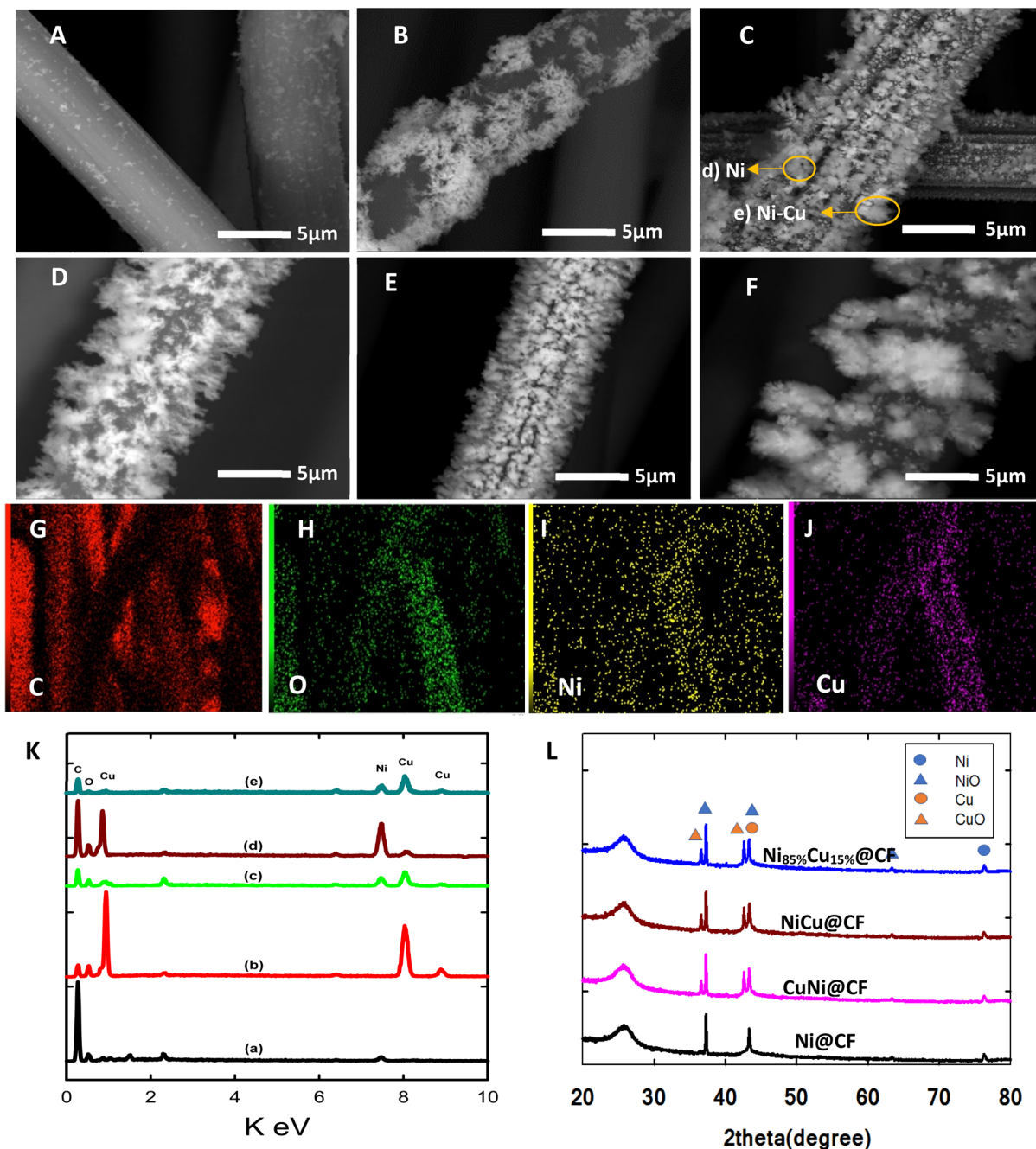


Fig. 1 SEM images of (A) Ni@CF, (B) Cu@CF, (C) Ni₈₅Cu₁₅@CF, (D) NiCu@CF, (E) Ni₅₀Cu₅₀@CF, and (F) Ni₂₅Cu₇₅@CF electrodes. (G) The corresponding-colored mapping EDX for NiCu@CF electrode, carbon (G), oxygen (H), nickel (I), and copper (J). (K) The corresponding spot EDX analysis of (a) Ni@CF, (b) Cu@CF, (c) NiCu@CF, (d and e) Ni₈₅Cu₁₅@CF electrodes, and (L) XRD patterns of Ni@CF, Cu@CF, NiCu@CF, CuNi@CF, and Ni₈₅Cu₁₅@CF electrodes. The same Ni loadings have been used in all electrodes' preparation.

EDX of Ni₈₅Cu₁₅@CF at two spots (Fig. 1K(d and e)) in which d spot contained mainly Ni and e spot contained both Ni and Cu hence higher electrochemically active surface sites are expected for NiCu@CF electrode compared with all other electrodes, see forward Fig. 2 and 5E.

It was quite hard to estimate the average particle size using the SEM technique, thus we used Image J software to estimate the average particle size of the modified CF with bimetallic Cu and Ni with various Cu/Ni ratios. SEM images, as shown in

Fig. 1(A–F), are used to estimate the average particle size. It's clear that by increasing the percent of Cu in the bath the average particle size also increased due to increasing the length of the dendritic structure.

Fig. 1K shows the respective EDX analyses for the above-mentioned electrodes. Ni@CF electrode exhibits peaks for carbon (from the CF substrate), oxygen (from CF and deposited oxygenated metal particles), and Ni elements. While the Cu@CF electrode shows peaks for carbon, oxygen, and Cu elements. On



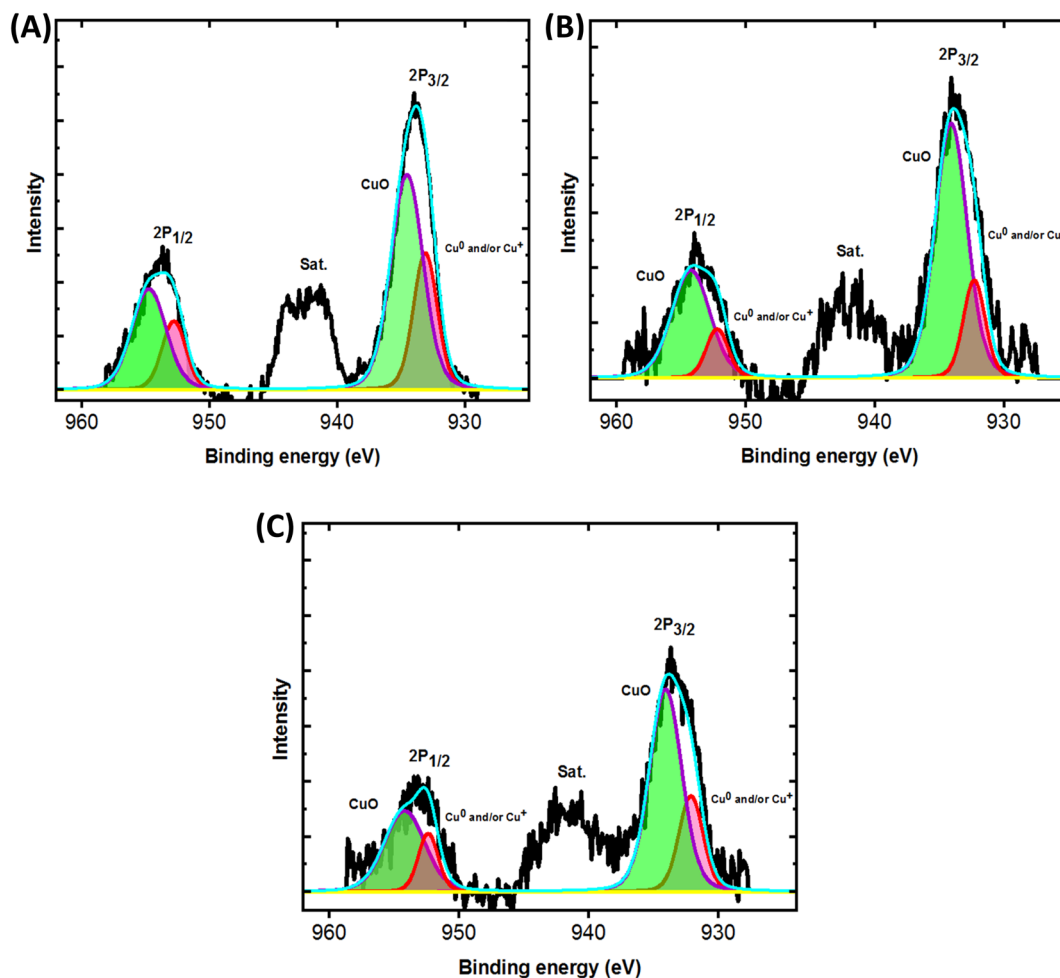


Fig. 2 High resolution XPS spectrum of Cu 2P: (A) Cu@CF, (B) NiCu@CF, and (C) Ni_{85%}Cu_{15%}@CF.

the other hand, the bimetallic decorated CF electrodes (*e.g.*, NiCu@CF, and Ni_{85%}Cu_{15%}@CF) show peaks for carbon, oxygen, nickel, and copper, ensuring the successful co-electrodeposition of both Cu and Ni particles atop of the CF. It is important to mention here that in the co-deposition technique, the electro-deposition of Cu is faster than Ni (thermodynamically). So, reducing the Cu concentration in the deposition bath makes the Ni content increase which reflects a higher ability to oxidize glycerol because Ni is more active than Cu as can be seen in Fig. 3B and 4B as well as the uniform distribution of the Ni and Cu is confirmed by the mapping EDX analysis for NiCu@CF electrode as depicted from Fig. 1(G and L).

The crystallographic orientations of the as-prepared electrodes were further investigated using XRD, as shown in Fig. 1H. It presents the XRD patterns of Ni@CF, NiCu@CF (Ni top layer), CuNi@CF (Cu top layer), and Ni_{85%}Cu_{15%}@CF electrodes. All the electrodes exhibited broad peaks at 2θ around 25° , and 43° attributed to the CF support.⁴⁹ On the other hand, all the electrodes exhibited an additional 4 diffraction peaks which could be attributed to cubic Ni, NiO, and CuO which matched with their reference codes (01-070-1849, 01-073-1519, and 01-078-0428). Table 1 shows the change in lattice constant ($a_{\text{exp}} = d\sqrt{h^2 + k^2 + l^2}$) where a_{exp} lattice constant,

d interplanar spacing and hkl Miller indices, lattice volume (lattice volume = $(a_{\text{exp}})^3$), and average crystal size calculated from Scherrer equation for all the prepared catalysts.^{41,50,51} As can be seen in Table 1, the addition of Cu to Ni causes increase in each of lattice constant, lattice volume, and crystal size due to the ionic radius of Cu, which is equal to 0.73 Å, is larger than the ionic radius of Ni which is equal to 0.69 Å.⁵¹

Furthermore, XPS was used to acquire more information about composition of the surface and oxidation state of the electrodeposited Cu. Fig. 2 shows high resolution XPS Cu 2p for Cu@CF, NiCu@CF and Ni_{85%}Cu_{15%}@CF electrodes. The oxidation state of Cu in the aforementioned electrodes is found to be (0 and/or +1 and +2) according to XPS analysis (deconvoluted peaks) where, the bands at the binding energies 932.7 eV and 952.7 eV are corresponding to Cu⁰ and/or Cu¹⁺ and at the binding energies 934.4 eV and 954.7 eV are attributed to Cu²⁺ that also confirmed by the presence of the satellite peaks as displayed in Fig. 2. Additionally, the Cu 2p peaks are shifted to lower binding energies in the presence of Ni (NiCu@CF and Ni_{85%}Cu_{15%}@CF) by ~ 0.4 eV which suggests the interaction between Ni and Cu particles consequently better activity of the prepared catalysts observed as can be seen in the Fig. 3.



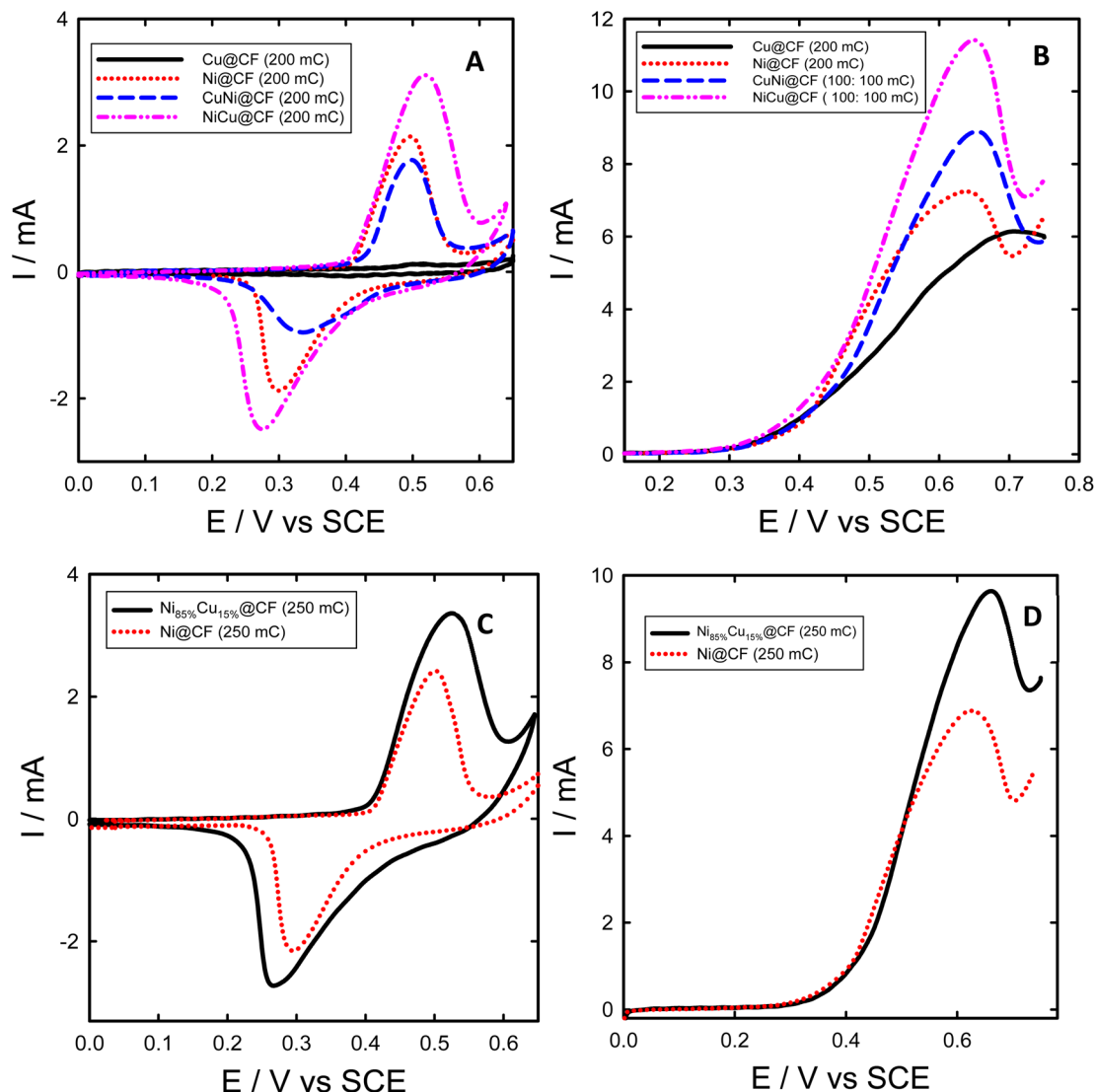


Fig. 3 CVs of NiCu@CF (dash-dot-dot), CuNi@CF (long dash), Ni@CF (dotted), and Cu@CF (solid) electrodes in 0.2 M NaOH solution without glycerol (A) and their corresponding LSVs in presence of 8 mM glycerol (B). CVs of Ni@CF, and Ni_{85%}Cu_{15%}@CF electrodes in 0.2 M NaOH without glycerol (C), and their corresponding LSVs in presence of 8 mM glycerol (D) all the measurements performed with potential scan rate of 10 mV s⁻¹.

3.2. Glycerol electrooxidation

CV technique was used to study electrochemical behavior and the performance of the as-prepared electrodes towards glycerol electrooxidation, as displayed in Fig. 3. Fig. 3(A and C) shows characteristic CVs obtained for Ni@CF, Cu@CF, NiCu@CF, CuNi@CF, and Ni_{85%}Cu_{15%}@CF electrodes in 0.2 M NaOH solution. As seen in Fig. 3(A and C), all the studied electrodes except the Cu@CF exhibited a redox peak couple at around 0.5 V vs. SCE which is attributed to the β -Ni(OH)₂/β-NiOOH transformation.^{21,41,52} Indeed, Cu@CF did not show any noticeable features under the measuring conditions. In the case of electrodes consisting of Ni and Cu, during the forward scan firstly Ni is converted to Ni(OH)₂ at lower potential and by raising the potential, the surface became a mixture of NiOOH and (Cu(OH)₂ or CuO) where the later one can also be converted to Cu(III).⁴¹

Interestingly, NiCu@CF and Ni_{85%}Cu_{15%}@CF electrodes showed stronger β-Ni(OH)₂/β-NiOOH transformation peaks compared to Ni@CF electrode of the same loadings (60.81 and 76.02 μg), where 1.5 and 1.4 times higher β-Ni(OH)₂/β-NiOOH peak intensities were obtained for NiCu@CF and Ni_{85%}Cu_{15%}@CF electrodes, respectively, suggesting much higher active surface concentration of Ni in the binary Cu–Ni electrodes. In other words, the bimetallic Cu–Ni decorated CF electrodes have higher Ni active surface sites compared to the decorated CF with pure Ni particles, despite their large average particle size. This increase of the surface-active sites could be attributed to the bimetallic dendrite-like structure as demonstrated from their respective SEM images (see Fig. 1(A–F)).

We estimated the surface concentration (Γ) of all electrodes using the following equation (eqn (1)):⁵²



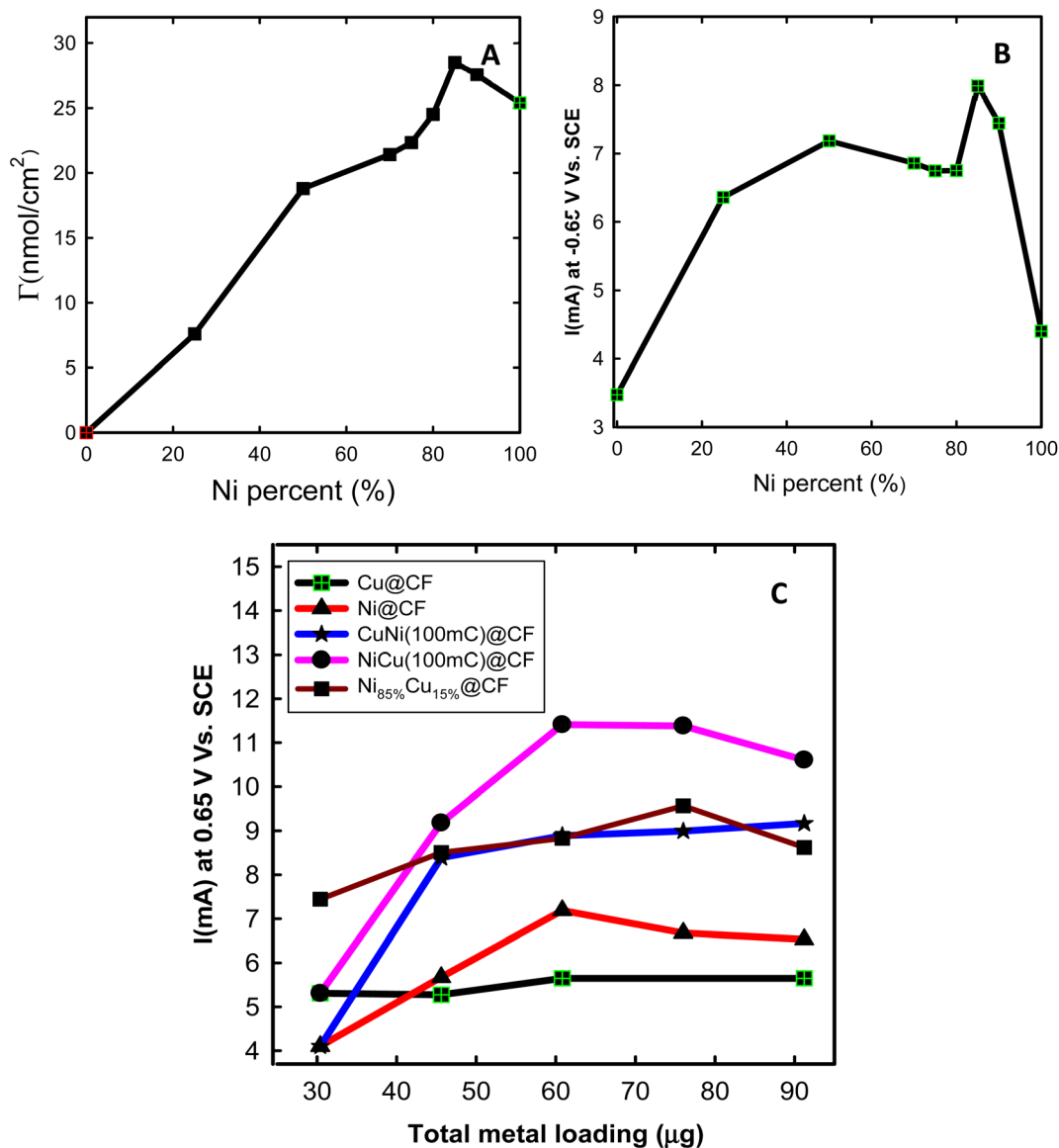


Fig. 4 (A) Surface concentration (Γ) of the prepared electrodes vs. Ni percent (mol%) in the deposition bath. (B) The variation of glycerol electrooxidation current recorded at 0.65 V (vs. SCE) with different Ni percent (mol%). (C) The variation of glycerol electrooxidation current at 0.65 V (vs. SCE) for various metal loadings in 0.2 M NaOH solution containing 8 mM glycerol of NiCu@CF, CuNi@CF, Ni@CF, Cu@CF, and Ni_{85%}Cu_{15%}@CF electrodes.

Table 1 The lattice constant, lattice volume, and crystal size for all the prepared catalysts

Factor	Composition	Ni@CF	NiCu@CF	CuNi@CF	Ni ₈₅ Cu ₁₅ @CF
Lattice constant (\AA)	NiO	4.155	4.156	4.157	4.162
	CuO	—	4.242	4.241	4.240
	Ni	3.528	3.528	3.529	3.530
Lattice volume (\AA^3)	NiO	71.743	71.809	71.823	72.101
	CuO	—	76.327	76.273	76.225
	Ni	43.898	43.912	43.950	43.987
Crystal size (nm)	NiO	37.8	38.1	38.9	40
	CuO	—	51.0	50.9	50.0
	Ni	24.1	24.1	25.0	26.5



$$\Gamma = Q/nFA \quad (1)$$

where Γ is the surface concentration of the active Ni(OH)/NiOOH (mol cm^{-2}), Q is the charge consumed during Ni(OH)/Ni(OOH) transformation (Coulomb (C)), taking $n = 1$, F is Faraday's constant ($F = 96\,500 \text{ C mol}^{-1}$), and A is the geometrical surface area of the working electrode (cm^2).³² The Γ was found to be 34, 12, 63, and 112 nmol cm^{-2} for Ni@CF, CuNi@CF, Ni_{85%}Cu_{15%}@CF, and NiCu@CF electrodes at the same Ni loading 60.8 μg , respectively. It is clear that further modification of Ni with Cu (sequential or co-deposition approach) resulted in a significant increase in the active surface species and NiCu@CF is the best electrode compared with the other electrodes which raise Γ by about 3.3 times over than Ni@CF electrode that may be due to the uniform homogeneous distribution of the Ni on the surface of the dendritic Cu (using sequential deposition technique).

Electrocatalytic activity of Ni@CF, Cu@CF, CuNi@CF, Ni_{85%}-Cu_{15%}@CF, and NiCu@CF electrodes were examined in 0.2 M NaOH solution containing 8 mM glycerol and the data is presented in Fig. 3. As obviously seen in Fig. 3(B and D), both NiCu@CF and Ni_{85%}Cu_{15%}@CF electrodes showed superior activity for glycerol oxidation compared to the Ni@CF, and pure Cu@CF electrodes. These enhancements are demonstrated by the large increase of the obtained glycerol oxidation peak current at the bimetallic electrodes compared to the single metal modified electrodes which increased by about 1.6 and 1.4 times in case of NiCu@CF and Ni_{85%}Cu_{15%}@CF respectively over than Ni@CF. This might be attributed to their unique dendritic-like structures providing more surface-active sites. By subtract the peak current of Ni oxidation in 0.2 M NaOH solution from the peak current of glycerol oxidation which obtained at the same potential scan rate (10 mV s^{-1}), we can figure out that enhancement in oxidation currents of glycerol is due to the increase of the oxidation peak of Ni or due to the synergistic effect between Ni and Cu particles in the dendritic-like structures. The enhancement after subtracting is found to be in the same ratios obtained previously that means the enhancement is due to the synergistic effect between Ni and Cu particles which is in agreement with the XPS analysis.

3.3. Optimization of the electrodeposition parameters

Different deposition parameters (*e.g.*, Cu:Ni ratio, loadings, *etc.*) have been optimized to prepare electrodes with the highest possible performance for glycerol electrooxidation.

3.3.1 Cu/Ni ratio effect. NiCu@CF electrodes with various Ni amounts were prepared *via* introducing various Ni concentrations in the Cu deposition bath and keeping the total concentration as 4 mM and then their respective CVs in 0.2 M NaOH solution in the presence and absence of glycerol were measured to track down the effect of Cu/Ni ratios. Fig. 4(A and B) shows the variation of the surface concentration (Γ , nmol cm^{-2}) and the obtained electrooxidation of glycerol current at 0.65 V as a function of the introduced Ni amount (mol%). As seen in Fig. 4A, the Ni surface concentration is increased with increasing the percent of Ni in the deposition bath to a certain value ($\sim 85\%$) and beyond this value, it starts to decrease. The

same trend is observed for the obtained glycerol electro-oxidation current, as shown in Fig. 4B.

3.3.2 Effect of Cu and Ni loadings. Effect of Ni and Cu loadings was further studied for NiCu@CF, CuNi@CF, and Ni_{85%}Cu_{15%}@CF electrodes to obtain the electrode with high activity and catalyst utilization for glycerol electrooxidation. To study the effect of Ni loading, Cu loading was kept constant at 23.9 μg . The same was done to study the effect of Cu loading, where Ni loading was kept constant at 30.4 μg . As shown in Fig. 4C, the obtained glycerol electrooxidation current at 0.65 V for all the investigated electrodes is gradually increased with the total metal loadings and reaches a plateau at certain metal loading (60.8 μg). Indeed, the catalytic activity of the as-prepared electrodes toward glycerol oxidation is in the order of NiCu@CF > Ni_{85%}Cu_{15%}@CF > CuNi@CF > Ni@CF > Cu@CF at all studied loading ranges. For example, NiCu@CF electrode shows ~ 1.6 times higher glycerol oxidation current compared to the single metal modified CF electrodes (Ni@CF and Cu@CF) with a total metal loading of 60.8 μg . While in case of Ni_{85%}-Cu_{15%}@CF electrode, which is synthesized *via* the co-deposition approach showed only 1.3 times higher activity as compared to Ni@CF electrode with the same metal loading. This may be attributed to the homogeneous Ni distribution atop of the Cu surface. So, Γ increases from 34 to 65 and 112 nmol cm^{-2} for Ni@CF, Ni_{85%}Cu_{15%}@CF, and NiCu@CF electrodes, respectively, with the same metal loading 60.8 μg .

3.4. Catalyst stability

Catalyst stability is one of the most important issues of fuel cell technology, thus the long-term stability of the optimized electrodes was next investigated *via* recording their chronoamperometric responses at 0.6 V for 2 h as shown in Fig. 5. As seen in this figure, all the investigated electrodes showed good stability under the measuring conditions. NiCu@CF and Ni_{85%}Cu_{15%}@CF electrodes supported higher oxidation currents with the lowest activity loss rates compared to the single metal modified CF electrodes. Clearly, Ni@CF and Cu@CF electrodes lost ~ 30 and 26%, respectively, from their initial performance after only 950 s while NiCu@CF and Ni_{85%}Cu_{15%}@CF lost only 8.5 and 8.9%, respectively, from their initial activities. Additionally, our prepared catalysts show better activity towards glycerol electrooxidation than most recent published papers, as compared in Table 2. SEM and mapping-EDX as can be seen from Fig. 6 are performed to figure out the structure and morphological stability for the best electrode (NiCu@CF) after 2 h of continuous operation. Fig. 6A–C shows three different points on the CF substrate to reveal that the dendritic morphology is still present and at the same time from the corresponding-colored mapping EDX Fig. 6D–G, it is clear that Ni and Cu are homogeneously distributed on the surface of the CF after long-term stability.

3.5. Electrochemical impedance spectroscopy (EIS) studies

The EIS was next measured to figure out the best prepared catalyst toward glycerol electro-oxidation. Nyquist plots of NiCu@CF, CuNi@CF, Ni@CF, Ni_{85%}Cu_{15%}@CF, and Cu@CF



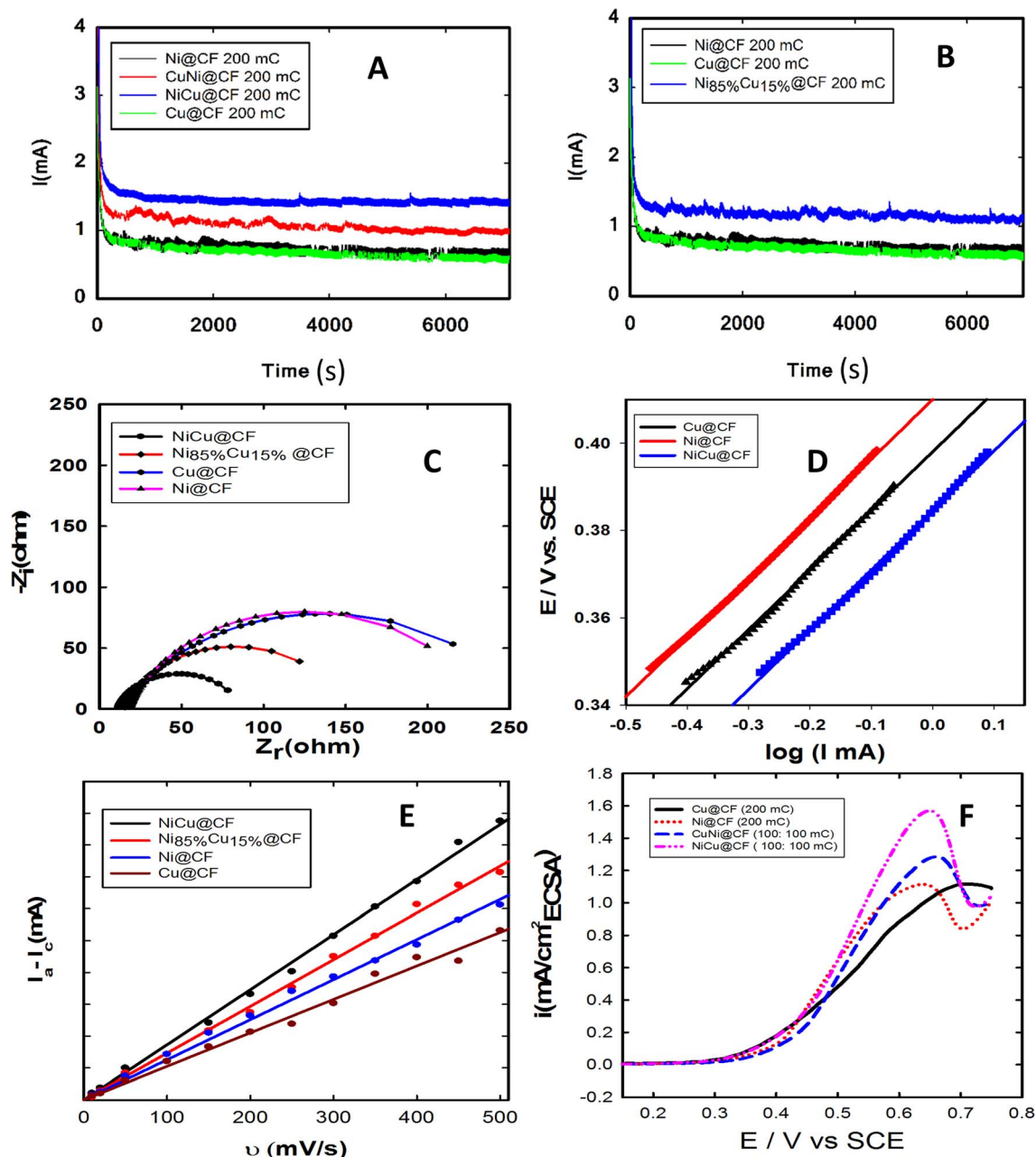


Fig. 5 Chronoamperometric curves measured at 0.6 V vs. SCE for (A) NiCu@CF, CuNi@CF, Ni@CF, and Cu@CF electrodes, and (B) Ni_{85%}-Cu_{15%}@CF, Cu@CF and Ni@CF electrodes. (C) Nyquist plots of NiCu@CF, CuNi@CF, Ni@CF, Ni_{85%}Cu_{15%}@CF, and Cu@CF electrodes measured at 0.4 V vs. SCE. (D) Tafel plots for Ni@CF, Cu@CF, and CuNi@CF at potential scan rate of 10 mV s⁻¹, and all of the measurements are carried out in 0.2 M NaOH solution containing 8 mM glycerol. (E) The estimation of electrochemical double layer capacitance (C_{dl}) for the different electrocatalysts by plotting the variation of I_c (capacitive current) vs. the potential scan rate in 0.2 M NaOH solution. (F) LSVs performed in 0.2 M NaOH solution containing 8 mM glycerol normalized to ECSA.

electrodes measured in 0.2 M NaOH solution containing 8 mM glycerol at 0.4 V vs. SCE displayed in Fig. 5C. As shown in Fig. 5C, the order of the charge transfer resistance (R_{ct}) is found to be NiCu@CF < Ni_{85%}Cu_{15%}@CF < Ni@CF < Cu@CF, suggesting that NiCu@CF is the best electrode which reduces R_{ct} from 214.2 Ohm in case of Ni@CF electrode to 75.5 Ohm. So, it has the faster glycerol electrooxidation kinetics at the bimetallic surfaces which may be due to Ni particles covering the surface of dendritic copper homogeneously. Thus, the surface

concentration of Ni is increased and at the same time the synergistic effect occurs between Ni and Cu. The obtained results were fitted using Randles equivalent circuit.

3.6. Tafel plots

Tafel plots were measured for Ni@CF, Cu@CF and NiCu@CF in 0.2 M NaOH solution containing 8 mM glycerol at potential scan rate of 10 mV s⁻¹, data are displayed in Fig. 5D which allow to



Table 2 A comparison of the electrocatalytic activities of various catalysts towards glycerol electrooxidation

Catalyst	[Glycerol] mol L ⁻¹	[NaOH] mol L ⁻¹	<i>I</i> (A g ⁻¹) at 1.6 V vs. RHE	Scan rate (mV s ⁻¹)	Ref.
CoNi@C	0.1	0.1	0.063	50	20
FeNi@C	0.1	0.1	0.052	50	20
FeCoNi@C	0.1	0.1	0.065	50	20
Ni@(CCE) ^b	0.1	1.0	24.372	50	41
NiCu@CCE	0.1	1.0	43.871	50	41
NiCo@CCE	0.1	1.0	116.990	50	41
Porous Cu/Cu ₂ O	0.1	0.1	6.6 at 1.5 V	10	56
20 mM Ni doped porous Cu/Cu ₂ O	0.1	0.1	6.423	10	57
Ni wire	0.1	1 M KOH	0.47	50	21
Ni wire/SW ^a	0.1	1 M KOH	1.5	50	21
Ni _{90%} Bi _{10%}	0.1	1 M KOH	360	50	58
NiCu@CF	0.008	0.2	140.892	10	This work
Ni _{85%} Cu _{15%} @CF	0.008	0.2	118.695	10	

^a (CCE) carbon ceramic electrode. ^b (SW) applying sinusoidal wave.

investigate the effect of addition of Cu on the kinetics of glycerol electrooxidation. Tafel slopes for Ni@CF, Cu@CF, and NiCu@CF were found to be 136, 135 and 132 mV dec⁻¹. The nominal similarity in Tafel slopes indicates that all the prepared electrodes have the same rate determining steps. Furthermore, the exchange current density (*i*₀) is estimated to be 0.27, 0.30, and 0.48 nA m⁻² for Ni@CF, Cu@CF and NiCu@CF electrodes, respectively, which further support the facile kinetics of the glycerol electrooxidation at NiCu@CF electrode.

3.7. Electrochemical active surface area (ECSA)

To figure out the enhancement in the activity simply due to increase in the number of active sites or by the action of the intrinsic activity. The intrinsic activity of the prepared

electrodes can be revealed from electrochemical active surface area (ECSA) normalization, see Fig. 5(E and F).^{53–55} To estimate the ECSA, the relation between the capacitive non-faradaic current density with the potential scan rate was used to estimate the ECSA using eqn (2) (ref. 54) for Ni@CF, Cu@CF, Ni_{85%}Cu_{15%}@CF, and NiCu@CF, see Fig. 5E.

$$\text{ECSA}(\text{cm}^2) = \frac{C_{\text{dl}}(\text{mF})}{C_s\left(\frac{\text{mF}}{\text{cm}^2}\right)} = \frac{(I_a - I_c)}{2\nu C_s} \quad (2)$$

where *C*_{dl} is the capacitive double layer ($C_{\text{dl}} = \frac{(I_a - I_c)}{2\nu}$), *I*_a is the anodic current and *I*_c is the cathodic current while *C*_s is

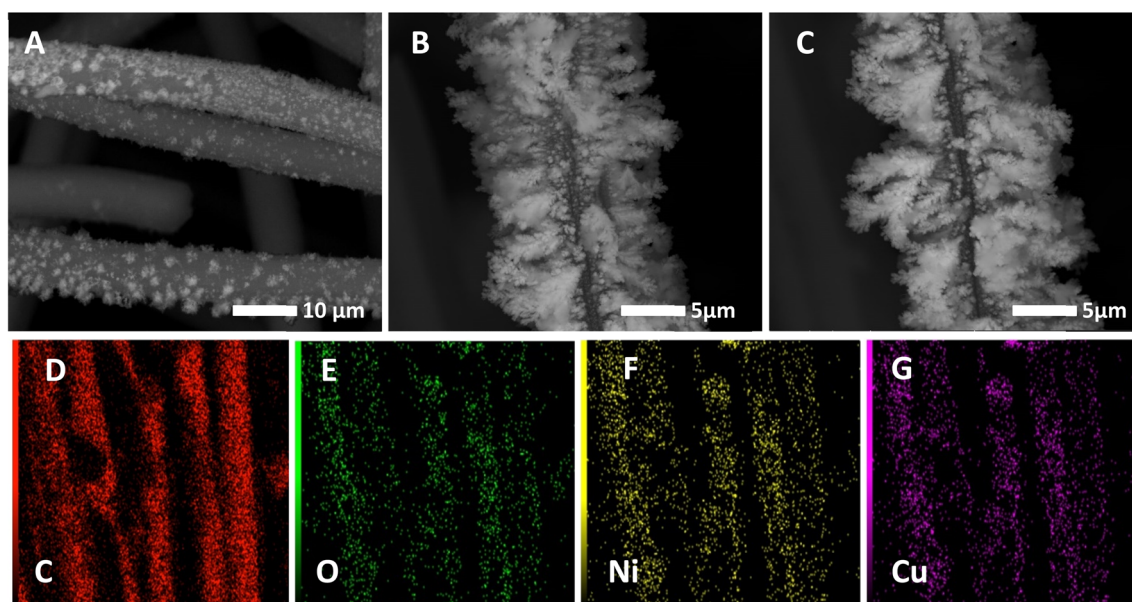


Fig. 6 SEM images for NiCu@CF electrode after 2 h of continuous operation at three different points on CF (A–C). The corresponding-colored mapping EDX analysis for: carbon (D), oxygen (E), nickel (F), and copper (G).



the specific surface area equal to 0.04 mF cm^{-2} in NaOH. The estimated ECSA found to be in order of $\text{NiCu@CF} > \text{Ni}_{85\%}\text{-Cu}_{15\%}\text{@CF} > \text{Ni@CF} > \text{Cu@CF}$ thus it is believed that NiCu@CF has the better electro-catalytic activity towards glycerol electro-oxidation. Moreover, NiCu@CF (sequential deposited) has the highest redox charge peak which further supports its catalytic activity as depicted from Fig. 3 and that in agreement with the idea of the better cover of Ni on the dendritic Cu which creates a large number of active sites. Also, Fig. 5F shows that after normalization the obtained current by ECSA, NiCu@CF is the best electrode. Thus, it is believed that the enhancement in the activity returns to the increase in the number of active sites as well as small increase in the intrinsic activity.

4. Conclusion

A novel catalyst was prepared by Cu metal insertion to Ni using sequential electrodeposition in which reverse deposition' order of Ni and Cu was investigated and at the same time different ratios of Ni and Cu in the deposition bath were investigated (co-deposition). In case of the sequential electrodeposition, the presence of Ni on the outermost layer (NiCu@CF electrode) was found to boost the performance by ~ 1.6 times higher than that at Ni@CF electrode with stability loss of only 8.5% after ~ 2 h of continuous operation as well as reducing the onset potential by about 63 mV. In case of the co-deposition, great enhancement in the activity and stability is achieved when the Ni to Cu ratio is equal to 85 : 15%. The electrocatalytic activity was ~ 1.4 times higher than that at Ni@CF electrode with a stability loss of only 8.9%. So, the electrode prepared *via* sequential deposition technique was found to be better than that prepared by the co-deposition one. This may be due to Ni particles covering the surface of dendritic copper in a uniform manner, in case of sequential technique. Thus, the surface concentration of Ni is increased from 34 to 63, and 112 nmol cm^{-2} for Ni@CF , $\text{Ni}_{85\%}\text{Cu}_{15\%}\text{@CF}$, and NiCu@CF electrodes, respectively, and a synergistic effect occurs between Ni and Cu. At the same time, electrode kinetics was enhanced towards glycerol electro-oxidation which reduces the charge transfer resistance from 214.2 to 75.5, and 134.9 Ohm for Ni@CF , NiCu@CF , and $\text{Ni}_{85\%}\text{Cu}_{15\%}\text{@CF}$ electrodes, respectively.

Conflicts of interest

There are no conflicts to declare.

References

- W. Tokarz, G. Lota, E. Frackowiak, P. Piela and A. Czerwinski, *Electrochim. Acta*, 2013, **98**, 94–103.
- J. R. C. Salgado, A. Anson, I. Tacchini, M. F. Montemor, E. R. Gonzalez, V. Paganin, M. A. Salvador, P. Ferreira, F. M. L. Figueiredo and M. G. S. Ferreira, *Int. J. Hydrogen Energy*, 2012, **38**, 910–920.
- M. S. Kim, B. Fang, N. K. Chaudhari, M. Song, T. S. Bae and J. S. Yu, *Electrochim. Acta*, 2010, **55**, 4543–4550.
- M. E. Ghaith, G. A. El-Nagar, M. G. Abd El-Moghny, H. H. Alalawy, M. E. El-Shakre and M. S. El-Deab, *Int. J. Hydrogen Energy*, 2020, **45**, 9658–9668.
- B. C. Ong, S. K. Kamarudin and S. Basri, *Int. J. Hydrogen Energy*, 2017, **42**, 10142–10157.
- M. S. Lashkenari, M. Ghorbani, N. Silakhori and H. Karimi-Maleh, *Mater. Chem. Phys.*, 2021, **262**, 124290.
- C. Qiu, R. Shang, Y. Xie, Y. Bu, C. Li and H. Ma, *Mater. Chem. Phys.*, 2010, **120**, 323–330.
- G. Li, B. Shi, Y. Gong, Y. Zhang, X. Wang, M. Guo and X. Lyu, *Mater. Chem. Phys.*, 2020, **243**, 122570.
- R. Ciriminna, C. Pina, M. Rossi and M. Pagliaro, *Eur. J. Lipid Sci. Technol.*, 2014, 1–22.
- C. A. G. Quispe, C. J. R. Coronado and J. A. Carvalho, *Renewable Sustainable Energy Rev.*, 2013, **27**, 475–493.
- B. T. X. Lam, M. Chiku, E. Higuchi and H. Inoue, *Adv. Nanopart.*, 2016, **05**, 60–66.
- A. N. Gerald, D. F. Da Silva, L. G. D. A. E Silva, E. V. Spinacé, A. O. Neto and M. C. Dos Santos, *J. Power Sources*, 2015, **293**, 823–830.
- H. H. Abdelhady, H. A. Elazab, E. M. Ewais, M. Saber and M. S. El-Deab, *Fuel*, 2020, **261**, 116481.
- Z. Zhang, L. Xin, J. Qi, D. J. Chadderton and W. Li, *Appl. Catal., B*, 2013, **136–137**, 29–39.
- A. Ashok and A. Kumar, *Int. J. Hydrogen Energy*, 2020, **46**, 4788–4797.
- C. Liu, M. Hirohara, T. Maekawa, R. Chang, T. Hayashi and C. Y. Chiang, *Appl. Catal., B*, 2020, **265**, 118543.
- M. Simões, S. Baranton and C. Coutanceau, *Appl. Catal., B*, 2010, **93**, 354–362.
- V. L. Oliveira, C. Morais, K. Servat, T. W. Napporn, P. Olivi, K. B. Kokoh and G. Tremiliosi-Filho, *Electrocatalysis*, 2015, **6**, 447–454.
- V. L. Oliveira, C. Morais, K. Servat, T. W. Napporn, G. Tremiliosi-Filho and K. B. Kokoh, *J. Electroanal. Chem.*, 2013, **703**, 56–62.
- V. L. Oliveira, C. Morais, K. Servat, T. W. Napporn, G. Tremiliosi-Filho and K. B. Kokoh, *Electrochim. Acta*, 2014, **117**, 255–262.
- M. S. E. Houache, E. Cossar, S. Ntais and E. A. Baranova, *J. Power Sources*, 2018, **375**, 310–319.
- Y. Wang, Y. Wang, J. Zang, L. Dong, H. Pan and Y. Yuan, *Electrochim. Acta*, 2013, **113**, 583–590.
- Z. Qi, H. Geng, X. Wang, C. Zhao, H. Ji, C. Zhang, J. Xu and Z. Zhang, *J. Power Sources*, 2011, **196**, 5823–5828.
- Y. Zhao, L. Fan, J. Ren and B. Hong, *Int. J. Hydrogen Energy*, 2014, **39**, 4544–4557.
- A. Papaderakis, N. Pliatsikas, C. Prochaska, K. M. Papazisi, S. P. Balomenou, D. Tsiplakides, P. Patsalas and S. Sotiropoulos, *Front. Chem.*, 2014, **2**, 1–11.
- D. Soundararajan, J. H. Park, K. H. Kim and J. M. Ko, *Curr. Appl. Phys.*, 2012, **12**, 854–859.
- Y. Hong, H. J. Kim, H. J. Lee, J. Kim and S. Il Choi, *Front. Chem.*, 2019, **7**, 1–9.
- S. Chemchoub, L. Oularbi, A. El Attar, S. A. Younssi, F. Bentiss, C. Jama and M. El Rhazi, *Mater. Chem. Phys.*, 2020, **250**, 123009.



- 29 M. R. Rizk and M. G. Abd El-Moghny, *Int. J. Hydrogen Energy*, 2021, **46**, 645–655.
- 30 M. R. Rizk, M. G. Abd El-Moghny, A. Mazhar, M. S. El-Deab and B. E. El-Anadoul, *Sustainable Energy Fuels*, 2021, **5**, 986–994.
- 31 M. Rizk, M. Abd El-Moghny and M. El-Deab, *J. Electrochem. Soc.*, 2020, **167**, 114505.
- 32 Y. Kang, W. Wang, Y. Pu, J. Li, D. Chai and Z. Lei, *Chem. Eng. J.*, 2017, **308**, 419–427.
- 33 R. M. A. Tehrani and S. Ab Ghani, *Electrochim. Acta*, 2012, **70**, 153–157.
- 34 S. Lee, H. J. Kim, S. M. Choi, M. H. Seo and W. B. Kim, *Appl. Catal., A*, 2012, **429–430**, 39–47.
- 35 M. Kapkowski, P. Bartczak, M. Korzec, R. Sitko, J. Szade, K. Balin, J. Lelątko and J. Polanski, *J. Catal.*, 2014, **319**, 110–118.
- 36 Y. Z. Su, Q. Z. Xu, Q. S. Zhong, C. J. Zhang, S. T. Shi and C. W. Xu, *Mater. Res. Bull.*, 2015, **64**, 301–305.
- 37 A. Y. Tsivadze, M. R. Tarasevich, V. A. Bogdanovskaya and M. R. Ehrenburg, *Dokl. Chem.*, 2008, **419**, 54–56.
- 38 E. Lohrasbi and M. Asgari, *Adv. Anal. Chem.*, 2015, **5**, 9–18.
- 39 M. A. Goda, M. G. Abd El-Moghny and M. S. El-Deab, *J. Electrochem. Soc.*, 2020, **167**, 064522.
- 40 A. M. Ahmed, S. Y. Sayed, G. A. El-Nagar, W. M. Morsi, M. S. El-Deab and B. E. El-Anadoul, *J. Electroanal. Chem.*, 2019, **835**, 313–323.
- 41 B. Habibi and N. Delnavaz, *RSC Adv.*, 2016, **6**, 31797–31806.
- 42 M. R. Rizk, M. G. Abd El-Moghny, H. H. Abdelhady, W. M. Ragheb, A. H. Mohamed, H. F. Fouad, M. Mohsen, A. S. Kamel and M. S. El-Deab, *Int. J. Hydrogen Energy*, 2022, **47**, 32145–32157.
- 43 G. A. El-Nagar, I. Derr, T. Kottakkat and C. Roth, *ECS Trans.*, 2017, **80**, 1013–1022.
- 44 I. Danaee, M. Jafarian, F. Forouzandeh, F. Gobal and M. G. Mahjani, *Int. J. Hydrogen Energy*, 2008, **33**, 4367–4376.
- 45 I. Danaee, M. Jafarian, F. Forouzandeh, F. Gobal and M. G. Mahjani, *Int. J. Hydrogen Energy*, 2009, **34**, 859–869.
- 46 C. Liu, M. Hirohara, T. Maekawa, R. Chang, T. Hayashi and C. Y. Chiang, *Appl. Catal., B*, 2020, **265**, 118543.
- 47 A. M. Abdelrahim, M. G. A. El-moghny, M. E. El-shakre and M. S. El-deab, *J. Energy Storage*, 2023, **57**, 106218.
- 48 A. M. Abdelrahim, M. G. Abd El-Moghny, M. E. El-Shakre and M. S. El-Deab, *Electrochim. Acta*, 2023, **440**, 141726.
- 49 M. S. El-Deab, G. A. El-Nagar, A. M. Mohammad and B. E. El-Anadoul, *J. Power Sources*, 2015, **286**, 504–509.
- 50 N. Benipal, J. Qi, Q. Liu and W. Li, *Appl. Catal., B*, 2017, **210**, 121–130.
- 51 S. K. Kumar, N. Murali, D. Parajuli, A. Ramakrishna, P. S. V. Rao and M. P. Rao, *Solid State Technol.*, 2020, **63**, 4069–4077.
- 52 M. M. Saleh, A. M. Ghonim and B. E. El-Anadoul, *Int. J. Biosens. Bioelectron.*, 2017, **2**, 45–53.
- 53 S. Nitopi, E. Bertheussen, S. B. Scott, X. Liu, A. K. Engstfeld, S. Horch, B. Seger, I. E. L. Stephens, K. Chan, C. Hahn, J. K. Nørskov, T. F. Jaramillo and I. Chorkendorff, *Chem. Rev.*, 2019, **119**, 7610–7672.
- 54 I. O. Baibars, M. G. Abd El-Moghny and M. S. El-Deab, *J. Environ. Chem. Eng.*, 2022, **10**, 2.
- 55 F. Dionigi, J. Zhu, Z. Zeng, T. Merzdorf, H. Sarodnik, M. Gliech, L. Pan, W. X. Li, J. Greeley and P. Strasser, *Angew. Chem., Int. Ed.*, 2021, **60**, 14446–14457.
- 56 M. R. Rizk, M. G. Abd El-Moghny and M. S. El-Deab, *J. Electrochem. Soc.*, 2020, **167**, 114505.
- 57 M. R. Rizk, M. G. Abd El-Moghny, G. A. El-Nagar, A. A. Mazhar and M. S. El-Deab, *ChemElectroChem*, 2020, **7**, 951–958.
- 58 M. S. E. Houache, K. Hughes, R. Safari, G. A. Botton and E. A. Baranova, *ACS Appl. Mater. Interfaces*, 2020, **12**, 15095–15107.

