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# Facile synthesis of defect-rich RuCu nanoflowers for efficient hydrogen evolution reaction in alkaline media†

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Developing high-performance electrocatalysts toward hydrogen evolution reaction (HER) in alkaline media is highly desirable for industrial applications in the field of water splitting but is still challenging. Herein, we successfully synthesized RuCu nanoflowers (NFs) with tunable atomic ratios using a facile wet chemistry method. The Ru<sub>3</sub>Cu NFs need only 55 mV to achieve a current density of 10 mA cm<sup>-2</sup>, which shows ideal durability with only 4 mV decay after 2000 cycles, largely outperforming the catalytic properties of commercial Pt/C. The Ru<sub>3</sub>Cu NFs comprise many nanosheets that can provide more active sites for HER. In addition, the introduction of Cu can modulate the electronic structure of Ru, facilitate water dissociation, and optimize H adsorption/desorption ability. Thus, the flower-like structure together with the proper incorporation of Cu boosts HER performance.

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## Introduction

Hydrogen production *via* electrolytic water splitting is considered a promising technology to meet the growing demands for sustainable and clean energy resources.<sup>1–4</sup> As the heart of scalable hydrogen generation, it is urgent to develop highly active and robust electrocatalysts to drive the hydrogen evolution reaction (HER) on a practical scale.<sup>5</sup> Generally speaking, the reaction rate of HER in an acidic solution is significantly higher than that in an alkaline solution. However, water splitting in acidic media is technologically and commercially hindered by the lack of efficient and low-cost electrocatalysts for oxygen evolution reaction (OER) at the anode electrode.<sup>6</sup> Considering that OER is more feasible in alkaline media, it is important to investigate HER under alkaline conditions for further industrial applications.

Pt-based nanocrystals have been theoretically and experimentally proven to be the most promising HER electrocatalyst owing to their proper H adsorption/desorption energy and low overpotential.<sup>7–10</sup> Unfortunately, because of the large kinetic energy barrier for the water dissociation step, their HER activity in alkaline media is approximately 2 to 3 orders of magnitude lower than that in acidic solution.<sup>11,12</sup> Moreover, the scarcity and high cost of Pt-based catalysts further limit their large-scale commercial applications, which arouses unprecedented research interests in developing economical and effective Pt-free electrocatalysts. Non-Pt-based catalysts, such as Mo-, Co-, Ni- and Cu-based nanocrystals, have been extensively reported,<sup>13–16</sup> but they are still inadequate to meet the urgent demands of high activity and durability.

Recently, Ru-based nanocrystals have received great attention as alternative electrocatalysts for HER owing to their moderate hydrogen bond strength (~65 kcal mol<sup>-1</sup>) and robust anticorrosion ability.<sup>17,18</sup> However, the weak hydrolytic capacity of Ru is still a major obstacle to limiting its intrinsic activity in alkaline solution. In addition, the strong absorption of OH\* species can severely block the adsorption/desorption of H\* on Ru.<sup>19,20</sup> To overcome these obstacles, tremendous efforts have been devoted, including structure design, shape control and composition optimization.<sup>21–25</sup> Among them, alloying Ru with a transition metal (*e.g.* Ni, Co, and Cu) offers a great opportunity to reduce the energy barrier for the water dissociation process, which exhibits remarkable enhancement in catalytic activity toward HER.<sup>26–28</sup> For example, incorporating Cu heteroatom into Ru can optimize the adsorption energy of H\* on the catalyst through the modulation of the electronic structure caused by the ligand effect, which can benefit the HER process.<sup>29,30</sup>

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Apart from alloy engineering, rational controlling the morphology of nanocrystals is another promising approach to tailor the catalytic properties.<sup>31–33</sup> Two-dimensional structure (2D) has drawn much attention owing to the advantages of providing a large surface, exposing more active sites and enhancing the diffusion of electrolytes, which can offer a high possibility to facilitate the catalytic performance.<sup>34–37</sup> In this regard, three-dimensional (3D) nanoflowers comprising the units of 2D nanosheet build blocks have emerged as promising candidates for advanced electrocatalysts owing to their integrated advantages of 3D and 2D structures, such as high porosity, large surface area and corrosion-resistance compared with their counterparts.<sup>38–40</sup> The highly opened flower-like structure facilitates the electron transfer among the reaction intermediates<sup>41,42</sup> and provides additional low-coordinated active sites at their edges, steps, and kinks, which can enhance the HER kinetics.<sup>41,42</sup>

Herein, we report a facile wet chemical method for synthesizing a class of defect-rich RuCu nanoflowers (denote as Ru<sub>x</sub>Cu NFs) to meet the demands of advanced HER catalysts. Owing to the enhanced active sites originating from the 3D structure and the optimized electronic structure of Ru modulated by incorporation of Cu, the as-synthesized Ru<sub>3</sub>Cu NFs show a low overpotential of 55 mV, a Tafel slope of 52.7 mV dec<sup>-1</sup> and excellent long-term durability. The DFT calculations further indicate that both enhanced water dissociation and H adsorption abilities help improve HER performance, making Ru<sub>3</sub>Cu NFs possible for further practical applications.

## Experimental section

### Materials and chemicals

Ruthenium(III) acetylacetonate (Ru(acac)<sub>3</sub>, 97%), copper(I) chloride (CuCl, 99%), polyvinylpyrrolidone (PVP, *M<sub>w</sub>* ≈ 29 000), L-ascorbic acid (C<sub>6</sub>H<sub>8</sub>O<sub>6</sub>, AA), citric acid (C<sub>6</sub>H<sub>8</sub>O<sub>7</sub>, CA), glucose (C<sub>6</sub>H<sub>12</sub>O<sub>6</sub>) and Nafion (5 wt%) were purchased from Sigma-Aldrich. Benzyl alcohol (C<sub>6</sub>H<sub>5</sub>CH<sub>2</sub>OH, AR), ethylene glycol (EG), ethanol, acetone and potassium hydroxide (KOH) were purchased from Sinopharm Chemical Reagent Co., Ltd. The commercial Pt/C (20 wt%) and Ru/C (5%) were purchased from Alfa Aesar. The deionized water (18.2 MΩ cm<sup>-1</sup>) used in all experiments was prepared by passing through an ultra-pure purification system. All chemicals and materials were used as received.

### Synthesis of Ru<sub>3</sub>Cu nanoflowers (NFs)

In a standard procedure, 12 mg of Ru(acac)<sub>3</sub>, 1 mg of CuCl, 15 mg of CA, 100 mg of PVP, and 5 mL of benzyl alcohol were mixed in a 20 mL vial under magnetic stirring at room temperature for 3 h. The homogeneous mixture was then heated up to 160 °C and maintained for 3 h in an oil bath before it cooled naturally. Finally, the final product was collected by centrifugation and washed with ethanol and acetone several times. The syntheses of the Ru<sub>2</sub>Cu NFs and Ru<sub>6</sub>Cu NFs were conducted under the same conditions except that the amount of Ru(acac)<sub>3</sub> was varied to 8 and 24 mg, respectively.

### Preparation of carbon-supported catalysts

In a standard preparation, carbon black (Vulcan XC-72) was dispersed into ethanol and sonicated for 0.5 h. The as-prepared Ru<sub>2</sub>Cu NFs, Ru<sub>3</sub>Cu NFs, and Ru<sub>6</sub>Cu NFs (20 wt%) were added to this dispersion. This mixture was further stirred overnight. After that, the resultant samples were precipitated out by centrifugation and washed three times with ethanol and water. Then, the products were dried under 60 °C overnight to obtain the final catalysts.

### Characterization

Transmission electron microscopy (TEM) images of the obtained samples were obtained using a HITACHI HT-7700 microscope operated at 100 kV. High-resolution transmission electron microscopy (HRTEM) was performed using an FEI Tecnai F20 G2 microscope operated at 300 kV. High-angle annular dark-field scanning TEM (HAADF-STEM) and energy dispersive X-ray (EDX) mapping analyses were obtained using an FEI Titan ChemiSTEM equipped with a probe-corrector and a Super-X EDX detector system. This microscope was operated at 200 kV with a probe current of 50 pA and a convergent angle of 21.4 mrad for illumination. The X-ray diffraction (XRD) patterns were recorded on a Miniflex600 X-ray diffractometer in a scan range of 30–90° at a scan rate of 10.0° per min. An X-ray photoelectron spectrometer (XPS) was performed on ESCALAB 250Xi (Thermo, U. K.).

### Electrochemical measurement

The electrochemical performances of the Ru<sub>2</sub>Cu NFs/C, Ru<sub>3</sub>Cu NFs/C, Ru<sub>6</sub>Cu NFs/C, commercial Pt/C and Ru/C were measured using a three-electrode cell with a CHI760E electrochemical analyzer (CH Instrument, Shanghai) with a glassy carbon rotating disk electrode (RDE, area: ~0.196 cm<sup>2</sup>), graphite rod, and Ag/AgCl electrode serving as the working, counter, and reference electrodes, respectively. All potentials were referenced to the reversible hydrogen electrode (RHE), and all polarization curves were corrected for the *i*R contribution within the cell. To make catalyst ink, 5 mg of the Ru<sub>2</sub>Cu NFs/C, Ru<sub>3</sub>Cu NFs/C, Ru<sub>6</sub>Cu NFs/C and the commercial Pt/C and Ru/C catalysts were dispersed in 5 mL of a mixed solvent and sonicated for 10 min. The solvent contained a mixture of de-ionized water, isopropanol, and 5% Nafion 117 solution at a volumetric ratio of 8 : 2 : 0.05. 30 μL of the catalyst ink was added onto the RDE and dried under the air flow for 30 min to make the working electrode. The loading amount of the Ru<sub>2</sub>Cu NFs/C, Ru<sub>3</sub>Cu NFs/C, Ru<sub>6</sub>Cu NFs/C and the commercial Pt/C and Ru/C catalysts on the RDE was determined to be ~15.3 μg<sub>Ru or Pt</sub> cm<sup>-2</sup>. The electrolyte for cyclic voltammetry (CV) measurement and linear scan voltammetry (LSV) test for hydrogen evolution reaction (HER) was measured in 0.1 M KOH solution purged with pure Ar. The CV measurement was conducted in an Ar-saturated 0.1 M KOH solution at room temperature with a scan rate of 50 mV s<sup>-1</sup>. Before the electrochemical measurement, the working electrode was cleaned *via* a steady-state cyclic voltammetry (CV) scan with 20 repetitions over the 0–1 V potential



range (*vs.* RHE) at a scan rate of  $50 \text{ mV s}^{-1}$ . Linear sweep voltammetry (LSV) was conducted in the range of  $-0.4$  to  $0.1 \text{ V}$  (*vs.* RHE) at a scan rate of  $10 \text{ mV s}^{-1}$  and a rotation speed of  $1600 \text{ rpm}$  to remove hydrogen gas bubbles formed on the catalyst surface. The electrochemically active surface area (ECSA) of the samples was estimated using the CV method in  $0.1 \text{ M KOH}$  at different scan rates of  $20, 40, 60, 80,$  and  $100 \text{ mV s}^{-1}$ . The electrochemical double-layer capacitance ( $C_{dl}$ ) is from linear fitting, and the slope is the  $C_{dl}$  value. The ECSA of the catalyst on RDE is estimated as follows:

$$\text{ECSA} = \frac{C_{dl}}{C_s} \times S,$$

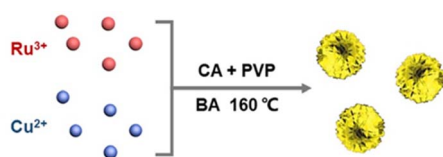
where  $C_s$  is the specific capacitance of the sample and the value of  $C_s$  is estimated to be  $0.04 \text{ mF cm}^{-2}$  in this study.  $S$  is the actual surface area of the electrode, which is equal to the geometric area of the glassy carbon electrode ( $0.196 \text{ cm}^2$ ). The durability tests were performed by applying cyclic potential sweeps between  $-0.2 \text{ V}$  and  $0.1 \text{ V}$  (*versus* reversible hydrogen electrode (*vs.* RHE)) at a scan rate of  $100 \text{ mV s}^{-1}$  for 2000 cycles. The chronoamperometry ( $i-t$ ) curves were measured at the potential with the current density being nearly  $10 \text{ mA cm}^{-2}$  for 12 h.

### Computational details

All the DFT calculations are performed in the framework of density functional theory using the projector-augmented plane-wave method, as implemented in the Vienna *ab initio* simulation package. The exchange–correlation potential was treated by the Perdew–Burke–Ernzerhof functional. The long-range van der Waals interaction is described by employing the DFT-D3 approach. The cut-off energy for the plane wave is set to  $450 \text{ eV}$ . The energy criterion is set to  $10^{-5} \text{ eV}$  in an iterative solution of the Kohn–Sham equation. A vacuum layer of  $15 \text{ \AA}$  is added perpendicular to the sheet to avoid artificial interaction between periodic images. The  $4 \times 4 \times 1$  Monkhorst–Pack  $k$ -point sampling grid was used for the Brillouin zone integration. All the structures were relaxed until the residual forces on the atoms decreased to less than  $0.02 \text{ eV \AA}^{-1}$ . The minimum energy path for each reaction step was confirmed using the nudged elastic band (NEB) method.

## Results and discussion

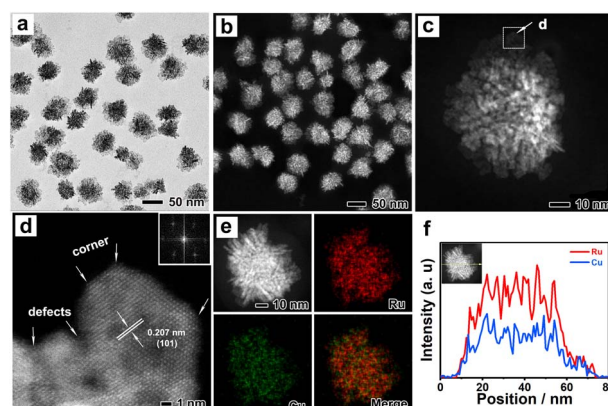
The typical synthesis procedure of  $\text{Ru}_3\text{Cu}$  NFs is shown in Scheme 1, where  $\text{Ru}_3\text{Cu}$  NFs can be prepared by applying a facile wet chemistry method using ruthenium(III)



**Scheme 1** Schematic illustration showing the synthetic process of the  $\text{Ru}_3\text{Cu}$  NFs.

acetylacetonate ( $\text{Ru}(\text{acac})_3$ ) and copper(I) chloride ( $\text{CuCl}$ ) as the metal precursors, benzyl alcohol as the solvent, citric acid (CA) as the reduction agent and polyvinylpyrrolidone (PVP) as the surfactant.  $\text{Ru}_x\text{Cu}$  NFs with different atomic ratios can be obtained by varying the amount of  $\text{Ru}(\text{acac})_3$  precursor fed in the synthesis. It can be found that the use of CA is crucial for the formation of unique flower-like nanostructures. As shown in Fig. S1,<sup>†</sup> only dendritic nanostructures could be obtained after replacing the CA with ascorbic acid (AA) or glucose. In addition, PVP acts as a capping agent and can effectively protect particles from aggregation. This demonstration can be supported by the TEM images; as depicted in Fig. S2a and b,<sup>†</sup> only aggregated particles can be obtained in the absence of PVP. Moreover, benzyl alcohol plays a key role in the formation of the ordered structure, replacing benzyl alcohol with ethylene glycol can lead to the formation of irregular nanoparticles (Fig. S2c and d<sup>†</sup>). Thus,  $\text{Ru}_3\text{Cu}$  NFs can be synthesized with the cooperative participation of CA, PVP and benzyl alcohol.

Fig. 1 shows the morphological, structural and compositional characterizations of  $\text{Ru}_3\text{Cu}$  nanocrystals prepared by employing a standard procedure. From transmission electron microscopy (TEM, Fig. 1a) and high-angle annular dark-field scanning transmission electron microscopy (HAADF-STEM, Fig. 1b), 3D flower-like nanostructures with a homogenous distribution and high productivity were clearly observed. The average size of the flower-like nanostructure is about  $44.7 \text{ nm}$  (Fig. S3<sup>†</sup>). To further uncover the morphological and structural features of the  $\text{Ru}_3\text{Cu}$  NFs, the aberration-corrected HAADF-STEM technique was carried out. As shown in Fig. 1c and S4,<sup>†</sup>  $\text{Ru}_3\text{Cu}$  NFs exhibit a highly opened 3D structure that comprises several ultrathin nanosheets as building blocks. Such a unique flower-like structure may endow  $\text{Ru}_3\text{Cu}$  nanocrystals with a large specific surface area and efficient mass transfer ability.<sup>30</sup> The magnified atomic-resolution HAADF-STEM observation (Fig. 1d) was performed to characterize the structure of sheet-like building blocks that focus on the white rectangle area, as illustrated in Fig. 1c. The lattice fringe of the nanosheet was



**Fig. 1** (a) TEM and (b) HAADF-STEM images of the  $\text{Ru}_3\text{Cu}$  NFs. (c) HAADF-STEM image of an individual  $\text{Ru}_3\text{Cu}$  NF. (d) Atomic-resolution HAADF-STEM image at higher magnification from a selected area in (c) and the corresponding FFT pattern. (e) EDX mapping images and (f) EDX line-scan profiles of the individual  $\text{Ru}_3\text{Cu}$  NF.



measured to be 0.207 nm, which can be indexed to the {101} plane of Ru (Fig. 1d). The corresponding fast-Fourier transformation (FFT, the inset of Fig. 1d) along the [0001] axis further indicates the hexagonal-close-packed (hcp) phase of Ru<sub>3</sub>Cu NFs and the fine crystallinity. Moreover, numerous defects, such as corners, kinks, and steps, could be observed along the edge of the nanoflowers, as marked in Fig. 1d and S4b,† which can provide additional active sites for HER. The elemental distribution of Ru and Cu in flower-like structures was characterized by energy-dispersive X-ray spectroscopy (EDX) analysis. As shown in Fig. 1e, both Ru and Cu were uniformly distributed throughout the nanoflowers, which agrees with the results of the EDX line-scan profiles (Fig. 1f). The atomic ratio of Ru/Cu was calculated to be 74.3/24.7 (Fig. S5†), which is similar to the result measured by inductively coupled plasma atomic emission spectroscopy (ICP-AES).

The crystal structure of the Ru<sub>3</sub>Cu NFs was characterized using the X-ray diffraction (XRD) technique. As shown in Fig. 2a, all diffraction peaks of the Ru<sub>3</sub>Cu NFs match well with those of hcp Ru (PDF # 06-0663). The lattice spacing of the (101) plane is measured to be 0.207 nm, which agrees with the HADDF-STEM results in Fig. 1d. XPS spectra were conducted to investigate the valence states and surface electron status of Ru<sub>3</sub>Cu NFs. As shown in Fig. 2b, the strong peaks located at 461.1 and 483.4 eV correspond to the Ru<sup>0</sup> 3p<sub>3/2</sub> and Ru<sup>0</sup> 3p<sub>1/2</sub>, indicating the majority of the metallic state in Ru. The weak peaks at 462.7 and 485.0 eV corresponding to Ru<sup>4+</sup> 3p<sub>3/2</sub> and Ru<sup>4+</sup> 3p<sub>1/2</sub> indicate that partial oxidation happened on the surface.<sup>43</sup> The XPS spectra of Cu 2p are shown in Fig. 2c. The peaks located at 932.5, 934.5, 952.4, and 954.5 eV can be attributed to the Cu<sup>0</sup> 2p<sub>3/2</sub>, Cu<sup>2+</sup> 2p<sub>3/2</sub>, Cu<sup>0</sup> 2p<sub>1/2</sub>, and Cu<sup>2+</sup> 2p<sub>1/2</sub>, respectively.<sup>28</sup> The peaks located at 942.1 eV are attributed to the satellite peaks. More impressively, the peaks of Ru in Ru<sub>3</sub>Cu NFs shifted to lower binding energy when compared with commercial Ru/C

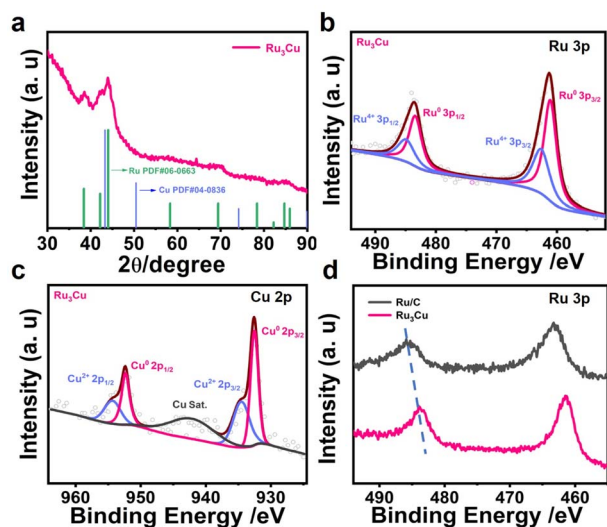


Fig. 2 (a) XRD pattern of the Ru<sub>3</sub>Cu NFs. (b) XPS spectra of Ru 3p. (c) Cu 4p orbitals of the Ru<sub>3</sub>Cu NFs. (d) XPS spectra of Ru 3p in the Ru/C and Ru<sub>3</sub>Cu NFs.

(Fig. 1d and S4†). It can be concluded that Cu transferred the electrons to Ru and made Ru an electron-rich state, which might have a positive effect on the catalytic activity towards the HER by optimizing the adsorption energy of intermediates.<sup>28,44</sup>

Ru<sub>2</sub>Cu and Ru<sub>6</sub>Cu NFs were synthesized through the same procedure, except for varying the amount of Ru precursor. As can be observed from the TEM and HADDF-STEM images in Fig. S7 and S8,† Ru<sub>2</sub>Cu NFs and Ru<sub>6</sub>Cu NFs show the same flower-like morphologies with average sizes of 44.5 nm and 43 nm, respectively. The XRD results further indicate that they are highly crystalline with an hcp Ru phase. Lattice fringes of the {002} Ru–Cu alloy with the interplanar distances of 0.217 and 0.215 nm can be observed in the Ru<sub>2</sub>Cu NFs and Ru<sub>6</sub>Cu NFs, respectively (Fig. S7f and S8f†). Both EDX mapping images and line-scan profiles confirm the homogeneous distribution of Ru and Cu throughout the nanoflowers, indicating the alloys of Ru and Cu (Fig. S5 and S6†).

To investigate the catalytic abilities of HER, Ru<sub>2</sub>Cu NFs, Ru<sub>3</sub>Cu NFs, and Ru<sub>6</sub>Cu NFs were first loaded onto carbon black (Vulcan X-72) for further measurement. The electrocatalytic measurements were carried out in a standard three-electrode system. The polarization curves of Ru–Cu NFs with different atomic ratios, commercial Ru/C and Pt/C were conducted in 0.1 M KOH solution. As shown in Fig. 3a, Ru<sub>3</sub>Cu NFs possess a low overpotential of 55 mV to achieve a current density of 10

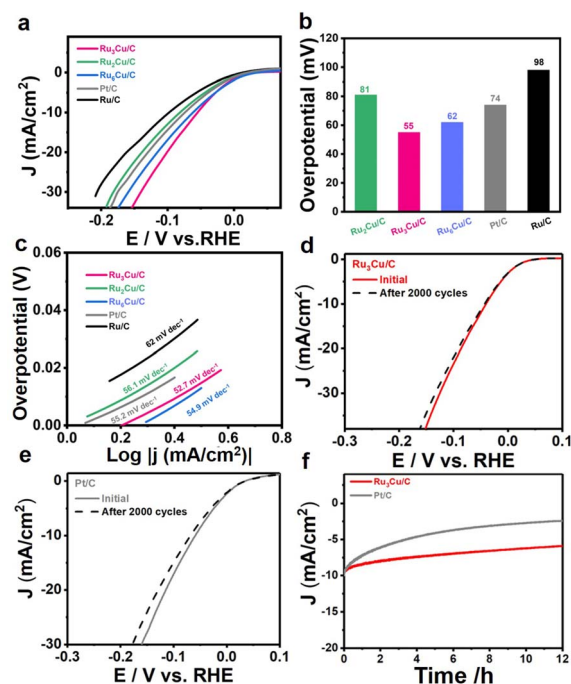


Fig. 3 (a) HER polarization curves of Ru<sub>2</sub>Cu NFs/C, Ru<sub>3</sub>Cu NFs/C, Ru<sub>6</sub>Cu NFs/C, Pt/C, and Ru/C in 0.1 M KOH solution. (b) Comparison of overpotentials at a current density of 10 mA cm<sup>-2</sup> with different catalysts. (c) Corresponding Tafel plots of different catalysts. (d) Polarization curves of Ru<sub>3</sub>Cu NFs/C and (e) Pt/C before and after 2000 cyclic voltammetry cycles between –0.2 V and 0.1 V (vs. RHE). (f) Chronoamperometry measurements of the Ru<sub>3</sub>Cu NFs/C and Pt/C at a potential of 10 mA cm<sup>-2</sup> for 12 h.



mA cm<sup>-2</sup>, which is much smaller than that of Ru<sub>2</sub>Cu NFs/C (81 mV), Ru<sub>6</sub>Cu NFs/C (62 mV), commercial Pt/C (74 mV), and Ru/C (98 mV) (Fig. 3b). Additionally, Ru<sub>3</sub>Cu NFs/C present the highest activity of 33.3 mA cm<sup>-2</sup> at the overpotential of -0.15 V vs. RHE, which is 1.5, 1.2, 1.4, and 1.8 times higher than that of Ru<sub>2</sub>Cu NFs/C (22.7 mA cm<sup>-2</sup>), Ru<sub>6</sub>Cu NFs/C (28.6 mA cm<sup>-2</sup>), Pt/C (24.3 mA cm<sup>-2</sup>), and Ru/C (18.4 mA cm<sup>-2</sup>) (Fig. S9†). To gain much insight into the reaction kinetics of HER, Tafel plots were derived from the polarization curves by fitting them with the Tafel equation. As shown in Fig. 3c, the Tafel slope of Ru<sub>3</sub>Cu NFs/C is 52.7 mV dec<sup>-1</sup> is much smaller than that of Ru<sub>2</sub>Cu NFs/C (56.1 mV dec<sup>-1</sup>), Ru<sub>6</sub>Cu NFs/C (54.9 mV dec<sup>-1</sup>), Pt/C (55.2 mV dec<sup>-1</sup>), and Ru/C (62 mV dec<sup>-1</sup>). This result indicates that the HER process on Ru<sub>3</sub>Cu NFs/C followed the Volmer–Heyrovsky pathway with water dissociation as the rate-determining step in an alkaline solution.<sup>45,46</sup> In addition, the double-layer capacitances (*C<sub>dl</sub>*) of the as-prepared catalysts were measured to evaluate the electrochemical active surface area (ECSA). As illustrated in Fig. S10 and S11,† the obtained specific capacitances for Ru<sub>2</sub>Cu NFs/C, Ru<sub>3</sub>Cu NFs/C, Ru<sub>6</sub>Cu NFs/C, and Pt/C were 20.3, 18.9, 16.4 and 12.2 mF cm<sup>-2</sup>, respectively, from which the ECSA value can be calculated. The larger ECSA value of Ru<sub>3</sub>Cu NFs/C (472.5 cm<sup>2</sup>) indicates that Ru<sub>3</sub>Cu NFs/C can expose more active sites than Pt/C (305 cm<sup>2</sup>). There is a volcano-like relationship toward HER performance among electrocatalysts with different Ru/Cu atomic ratios. This may be explained by the Sabatier principle, in which the proper surface electron structure states and H adsorption energy were provided in Ru<sub>3</sub>Cu NFs.<sup>44</sup> In a word, it can be speculated that the optimized catalytic performance of Ru<sub>3</sub>Cu NFs can be attributed to the opened structure and synergistic effect between Ru and Cu. However, the flower-like structure with a high density of defects and steps can provide additional active sites, and the highly opened structure is favorable for the rapid mass transfer process.<sup>41,47,48</sup> Obvious activity decay can be observed in Fig. S13† when substituting the flower-like structure with nanoclusters. The electron transfer occurred from Cu to Ru based on the negative shift of Ru 3p according to XPS spectra,<sup>29,41</sup> the ligand effect of Ru and Cu can optimize the H adsorption energy and facilitate the water dissociation process, thereby boosting HER performance.

Long-term durability is an important parameter for evaluating HER electrocatalysts. The stability of these five catalysts was evaluated through accelerated durability (ADT) for 2000 cycles between -0.2 and 0.1 V vs. RHE at a scan rate of 100 mV s<sup>-1</sup> in an Ar-saturated 0.1 M KOH solution. Fig. 3d shows the HER polarization curves of the Ru<sub>3</sub>Cu NFs/C before and after the test. There is no obvious change in HER polarization curves after 2000 cycles. The overpotential of Ru<sub>3</sub>Cu NFs/C at 10 mA cm<sup>-2</sup> gives only a slight increase of 4 mV in an alkaline solution, suggesting excellent stability. In contrast, Ru<sub>2</sub>Cu NFs/C, Ru<sub>6</sub>Cu NFs/C, Pt/C, and Ru/C perform extinguished decay with a larger overpotential increase of 7, 5, 9, and 8 mV, respectively (Fig. S12 and S14†). The long-term stability of the as-prepared Ru<sub>3</sub>Cu NFs/C was examined using chronoamperometry (*i-t*) measurements. As shown in Fig. 3f, Ru<sub>3</sub>Cu NFs/C catalysts maintain a more stable current density over 12 h, while the commercial

Pt/C decreased significantly, indicating the outperforming stability of Ru<sub>3</sub>Cu NFs/C. The morphology of Ru<sub>3</sub>Cu NFs/C and Pt/C after the durability test were further characterized by TEM, HRTEM, EDX mapping and spectrum analyses. As shown in Fig. S15,† Ru<sub>3</sub>Cu NFs/C still maintained the initial flower-like structure with good crystallinity, which is responsible for its outstanding durability. Moreover, EDX-spectrum analysis reveals that the Cu atoms were leached out in Ru<sub>3</sub>Cu NFs during the catalytic reaction (from 25.7 to 13.8), which may account for the decay of HER activity after the ADT. In comparison, commercial Pt/C exhibited obvious aggregation after 2000 cycles, as shown in Fig. S16,† resulting in the considerable decay of HER properties.

To reveal the intrinsic activity enhancement toward HER, first-principle density functional theory (DFT) calculations were conducted. The structure models of Ru<sub>3</sub>Cu, Ru and Pt are shown in Fig. 4a, where Ru<sub>3</sub>Cu (0001), Ru (0001), and Pt (111) were cleaved as the exposing surfaces to adsorb H<sub>2</sub>O. In alkaline media, the lack of protons makes the adsorption and cleavage of H<sub>2</sub>O to \*H and OH<sup>-</sup> on the surface of the catalyst a critical ability for the HER process. The barrier of water dissociation may lead to insufficient H–OH cleavage and hinder HER kinetics. Thus, HER in alkaline media is governed by the intrinsic ability of the catalyst to catalyze H\* to H<sub>2</sub> and the kinetics of water molecular dissociation. The free energy diagram of the water dissociation process is shown in Fig. 4b. After the incorporation of Cu, Ru<sub>3</sub>Cu NFs exhibit enhanced capability with a smaller energy barrier for water dissociation relative to Ru, which agrees with the results of the lower Tafel slopes, as depicted in Fig. 3c. Moreover, models of H adsorption on the surface of Ru<sub>3</sub>Cu, Ru and Pt are shown in Fig. 4c. The face-centered cubic (fcc) site on Ru<sub>3</sub>Cu, fcc site on Ru and fcc site on Pt were calculated to be preferable sites in each model. The calculated free energy of the above models is shown in

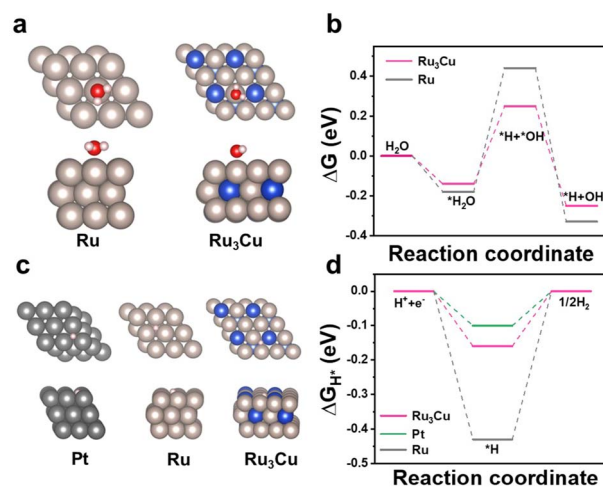


Fig. 4 Density functional theory calculation. (a) The models of adsorption H<sub>2</sub>O on Ru and Ru<sub>3</sub>Cu. (b) Free-energy diagram of the water dissociation process on Ru<sub>3</sub>Cu and Ru. (c) The models of adsorption of H on the surface of Pt, Ru and Ru<sub>3</sub>Cu. (d) Adsorption-free energies of H on Ru<sub>3</sub>Cu, Pt and Ru.



Fig. 4b. The  $\Delta G_{H^*}$  on Ru<sub>3</sub>Cu, Ru and Pt is calculated to be  $-0.16$ ,  $-0.43$  and  $-0.1$  eV, respectively. The  $\Delta G_{H^*}$  value of Ru<sub>3</sub>Cu NFs is close to zero, which is comparable to that of Pt, indicating the fast electron–proton transfer procedure and hydrogen desorption process. Thus, the facilitated water dissociation ability together with the optimized H adsorption/desorption ability contributes to outperforming HER performance.

## Conclusions

In summary, we successfully synthesized a series of RuCu NFs by applying a facile solvothermal method. The contents of Ru and Cu can be easily modulated by varying the amounts of salt precursors. It is indicated that the RuCu NFs comprise various thin nanosheets, leading to additional active sites for HER. The incorporation of Cu significantly modified the electronic structure of Ru. Surprisingly, the overpotential of Ru<sub>3</sub>Cu NFs at 10 mA cm<sup>-2</sup> reaches only 55 mV with a Tafel slope of 52.7 mV dec<sup>-1</sup>, which is superior to other Ru–Cu NFs, Ru/C and Pt/C. In addition, the Ru–Cu NFs exhibited enhanced stability for HER relative to Pt/C owing to their stable flower-like structure. DFT calculations indicate that both the facilitated water dissociation and appropriate H adsorption/desorption abilities are responsible for the outstanding catalytic performance. These findings provide a promising catalyst for HER on Ru-based catalysts, which may be applied to industrial production. This study provides a facile approach for the synthesis of 3D open nanostructures and a typical example of designing advanced Ru-based HER electrocatalysts with improved performance.

## Conflicts of interest

There are no conflicts to declare.

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