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Chloride-assisted synthesis of tellurene directly on SiO₂/Si substrates: growth mechanism, thermal properties, and device applications†

Yi-Hsun Chan,^a Che-Yi Lin,^b Yu-Chang Chou,^a Alice Chinghsuan Chang,^g
Yen-Fu Lin^{bdef} and Yu-Ze Chen^{ib*ac}

The new single-element two-dimensional materials in the post-graphene era, such as phosphorene and silicene, are emerging as alternative next-generation materials due to their remarkable physical properties. However, the vulnerability of silicene and phosphorene to the air hinders their practical applications. Recently, a new kind of single-element two-dimensional material, namely tellurene, exhibiting outstanding electrical mobility and superior long-term stability in the air, has been successfully synthesized. Here, we report that the addition of chloride (TeCl₄) to TeO₂ is conducive to the lateral growth of tellurium on the SiO₂/Si substrate. The resulting tellurene flakes display outstanding structure stability over 30 days under ambient exposure. Furthermore, the applicability of SThM (scanning thermal microscopy) to investigate the thermal properties of tellurene is presented for the first time, revealing its abnormal thermal conductance across grain boundaries and a thickness-dependent thermal transport behavior. Finally, tellurene-based FETs (field effect transistors) were fabricated, which displayed temperature-dependent I_{ds} that decreased with decreasing temperature rather than metallic transport features, which implies thermal activation of a semiconductor and confirms that the transport follows the 2D variable-range hopping (VRH) model.

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Introduction

In 2005, for the first time, scientists successfully exfoliated a single layer of graphite, known as graphene, which surprisingly exhibited superior properties in all aspects, such as outstanding catalytic ability,^{1,2} high transparency with high conductivity,³ and high electrical mobility at room temperature, as well as long-term stability in air.^{4,5} For over a decade, scientists have drawn their

attention away from graphene due to its major shortcoming of zero bandgap, which hinders its application in logics.⁶ Two dimensional transition metal dichalcogenides (TMDCs) have unfolded another vigorous research trend because of the inherent nature of their bandgap, which yields great potential for developing next-generation electronic devices. However, the current state-of-the-art TMDC-based transistors are barely satisfactory in terms of electron mobility, which is still far below that of the modern Si-based transistor. Recently, the widely and intensively investigated single-element two-dimensional materials made of graphene derivatives have exhibited intriguing photoelectric properties,⁷ energy-storage ability,⁸ and ultra-high electron mobility.⁹ Tellurium, which belongs to the chalcogens, possesses a unique chiral crystal structure, in which the atoms in a chain are bonded with the neighboring atoms by covalent bonding, and the two adjacent chains are attracted by van der Waals forces. Since 2004, the anisotropy in the atomic structure of Te has led to the limitation of its study and applications that are mainly based on the nanowire at the early stage of development.¹⁰ Tellurene (two-dimensional Te) has become one of the emerging two-dimensional materials because of its intriguing optical,^{11,12} magnetic^{13,14} and thermal properties.^{15–17} Recently, scientists have made great progress toward the synthesis of the lamellar structure of Te, called tellurene, by the substrate-free hydrothermal approach,^{18,19}

^a Department of Materials Science and Engineering, National Cheng Kung University, Tainan City, 70101, Taiwan. E-mail: yzchen@gs.ncku.edu.tw

^b Department of Physics, National Chung Hsing University, Taichung 40227, Taiwan

^c Program on Key Materials, Academy of Innovative Semiconductor and Sustainable Manufacturing, National Cheng Kung University, Tainan City, 70101, Taiwan

^d Department of Materials Science and Engineering, National Chung Hsing University, Taichung 40227, Taiwan

^e Institute of Nanoscience, National Chung Hsing University, Taichung 40227, Taiwan

^f i-Center for Advanced Science and Technology (i-CAST), National Chung Hsing University, Taichung 40227, Taiwan

^g Center for Measurement Standards, Industrial Technology Research Institute, Hsinchu, 30011, Taiwan

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low-temperature deposition *via* the PVD (physical vapor deposition) process on Si²⁰ and epitaxial growth on a single crystalline substrate *via* CVD (chemical vapor deposition).²¹ Y. Wang *et al.* have proposed a reliable solution for achieving the large-scale synthesis of tellurene with high crystallinity and also fabricated tellurene-based FETs (field effect transistors), displaying a high on/off ratio of 1×10^5 with excellent hole mobility ($\sim 700 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$). More strikingly, their most important finding is the ultra-long structural stability (~ 55 days) without any protective layer upon exposure to the air. However, the inert chemical activity and high anisotropy in the structure of tellurium render the process of obtaining lamellar tellurium with a much thinner thickness considerably time-consuming (up to 50 hours).¹⁸ X. Zhang *et al.* conducted the reduction of TeO₂ by simply using hydrogen to allow epitaxial deposition on the basal plane of mica. Additionally, through DFT calculations, they found that the intermediates played a critical role in the formation of tellurene. In addition, the usage of a single-crystalline substrate for the epitaxial growth of tellurene was inevitable.^{21–23} Hence, the synthesis of tellurene directly on an amorphous substrate for wider application still remains a challenge.

Herein, we propose a reliable method for synthesizing tellurium on arbitrary substrates and have discovered the influence of the ratio of TeCl₄/TeO₂ on the dimension of the resulting products, from one-dimensional to two-dimensional structure. Choosing precursors with low-melting points enables the direct synthesis of Te nanoflakes on arbitrary substrates, such as sapphire, mica and SiO₂/Si wafer. In addition, a systematic investigation of growth parameters during the formation of tellurene was conducted. As a result, the essential factors that determine the preferential growth direction of Te are proposed here, for the first time.

Results and discussion

Fig. 1(a) depicts the experimental setup of the CVD process, in which the precursors TeO₂/TeCl₄ were placed in the upstream region (750 °C), and the SiO₂/Si substrate was placed in the downstream region (750 °C) 30 cm away from the precursors for the deposition of Te. Once the process was completed, several single crystals were found loosely spread on the substrate (Fig. 1(b)), and the magnified images reflected light-pink hexagonal flakes (see the inset of Fig. 1(b)). Next, the thickness and dimension of the resulting materials were confirmed by AFM (atomic force microscope) to be around 10 nm (Fig. 1(c)). Note that if the thickness of tellurene was greater than 50 nm, it appeared the shiny golden, reflecting the metalloid innate. Since the discovery of graphene, Raman spectroscopy has been regarded as a helpful technique, especially, for the analysis of the vibration modes of two-dimensional materials and even the estimation of their thickness. Fig. 1(d) presents the typical spectrum of tellurene. The two featured peaks of the A₁ mode (123 cm⁻¹) and E₂ mode (140 cm⁻¹) were detected in accordance with previous reports, confirming the successful formation of tellurene. Remarkably, along with the increase in the thickness of tellurene, the signal from the substrate gradually vanished. Furthermore, it was discovered that the Raman peaks of the A₁ mode, which are associated with the vibration of the intrachain of Te, exhibited a redshift as the thickness of tellurene increased. This indicates that the van der Waals forces are influenced by thickness, resulting in the frequency change of the intra-chain vibrations.

In order to identify the atomic structure of tellurene, high-resolution TEM (transmission electron microscopy) was exploited. Prior to TEM observation, tellurene was capped with SiO₂ for

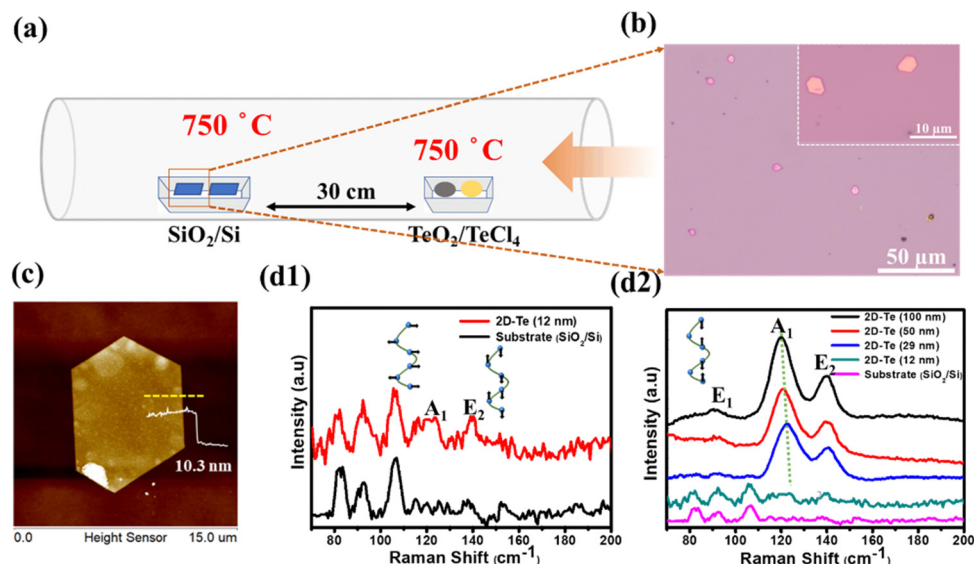


Fig. 1 Material characterization of tellurene. (a) Experimental setup of the CVD process, in which the hybrid precursors TeO₂/TeCl₄ are placed in the upper stream, and the SiO₂/Si substrate is placed 30 cm away from the precursors. (b) The optical image of the hexagonal tellurene flakes. (c) The AFM image confirming the thickness of tellurene to be around 10.3 nm. (d) The Raman spectrum of tellurene consists of peaks at 123 cm⁻¹ and 140 cm⁻¹, which correspond to the A₁ mode and E₂ mode, respectively.



protection from damage by the Ar^+ ions during the FIB (focused ion beam) process, as shown in Fig. 2(a). Fig. 2(b) displays the cross-sectional TEM image, clearly revealing the layered structure with a spacing of 3.9 Å, which matches the values in previous reports,^{22,24} and the preferential epitaxial direction corresponding to the (001) plane (Fig. 2(c)). Next, based on the compositional analysis by TEM-EDX, the element mapping showed that the signal Te was uniformly distributed, and the interface between Te and SiO_2 was sharply divided, as shown in Fig. 2(d1–d4). Considering the structural stability of tellurene in the air, the TEM sample of tellurene prepared by FIB was intentionally placed in the air for 30 days. After cross-sectional cutting by FIB, the edge of tellurene was fully exposed to the air, and the edge of the two-dimensional material confirmed that was susceptible to the air.²⁵ After that, TEM was applied for the estimation of its stability in the air. Significantly, the layered structure was still observed with no evident degradation, as shown in Fig. 2(e1–f2). The long duration of stability up to around 30 days evidently exceeds that of silicene which is stable for 48 min,¹⁹ and that of phosphorene, which is stable for 60 min.²⁶

To understand the role of the gas species during the synthesis of tellurene, the gas mixture of Ar/H_2 at varied ratios was investigated and the ratio of precursor $\text{TeO}_2/\text{TeCl}_4$ was set to 10 g/30 g. When pure Ar gas was used through the process, no tellurene flakes were observed, and only dark particles were spread on the surface (Fig. 3(a1)). Next, the hydrogen was introduced at a constant total flow flux of ($\text{Ar} + \text{H}_2 = 15 \text{ sccm}$), and it was found to participate in the reaction of $\text{TeO}_2/\text{TeCl}_4$. Once the hydrogen was introduced, hexagonal flakes appeared on the surface (Fig. 3(a2)). While increasing the amount of hydrogen to 8 sccm or 15 sccm, the color of the resulting flakes

obviously turned from yellow to pink (Fig. 3(a3 and a4)). Fig. 3(a5) plots the dependence of the lateral dimension and thickness of the flakes on the Ar/H_2 ratio. It is worth noting that the thickness of tellurene was inversely proportional to the proportion of hydrogen, which is possible due to the etching effect, as observed in previous works on graphene synthesis, proving that hydrogen could thin down the resultant thickness.²⁷ The thinnest value realized in this work was around 12 nm with pure hydrogen. Moreover, the total amount of hydrogen affected the dimension and thickness of tellurene (Fig. S1, ESI[†]). The process temperature was set to 750 °C and maintained for 30 min. When the flux of hydrogen was 5 sccm, there was no deposition of tellurene due to the insufficient carrier gas to deliver the Te vapor. When the flux was increased to 10 sccm, tellurene flakes with a thickness of 20 nm were achieved, but the dimension was irregular. When the flux was increased to 15 sccm, regular hexagonal tellurene flakes were realized. It is worth mentioning that when the flux was over 20 sccm, thicker tellurene flakes (~56.7 nm in thickness) were obtained, which can be attributed to the tendency of the excess Te vapor to nucleate on the existing Te flakes, resulting in the overlapping of multiple flakes.²⁸ Different from the previous works on tellurene synthesis through the simple reduction of TeO_2 with the assistance of hydrogen,²¹ in this work, low-melting-point TeCl_4 (melting point: 230 °C) is also chosen as another Te feedstock. It was found that the ratio of $\text{TeO}_2/\text{TeCl}_4$ played a decisive role in the growth mechanism during the transition from one-dimensional nanowire to two-dimensional epitaxial growth. Contrary to previous works,²¹ which solely use TeO_2 as the precursor for the reduction reaction using hydrogen, it was hard to synthesize tellurene thin films, but only less bulky Te with an irregular shape was found on the SiO_2/Si substrate

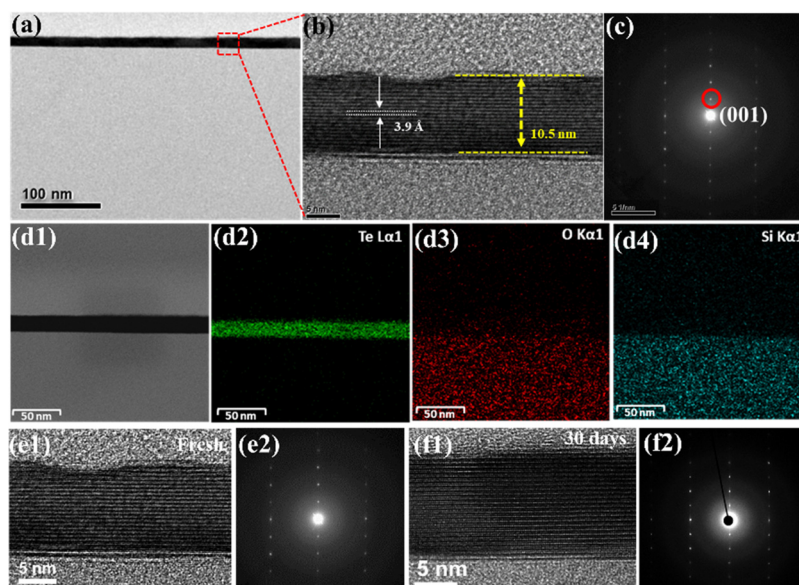


Fig. 2 TEM characterization and the structural stability test. (a) The cross-sectional image of tellurene. (b) The magnified view of 10.5 nm tellurene, revealing the layered structure with a lattice spacing of 3.9 Å. (c) SAED pattern. (d1–d4) The TEM-EDX mapping images of Te, O and Si, respectively. (e1) The cross-sectional image of fresh tellurene and (e2) the corresponding SAED pattern. (f1) The cross-sectional image of tellurene exposed to the air for over 30 days and (f2) the corresponding SAED pattern.



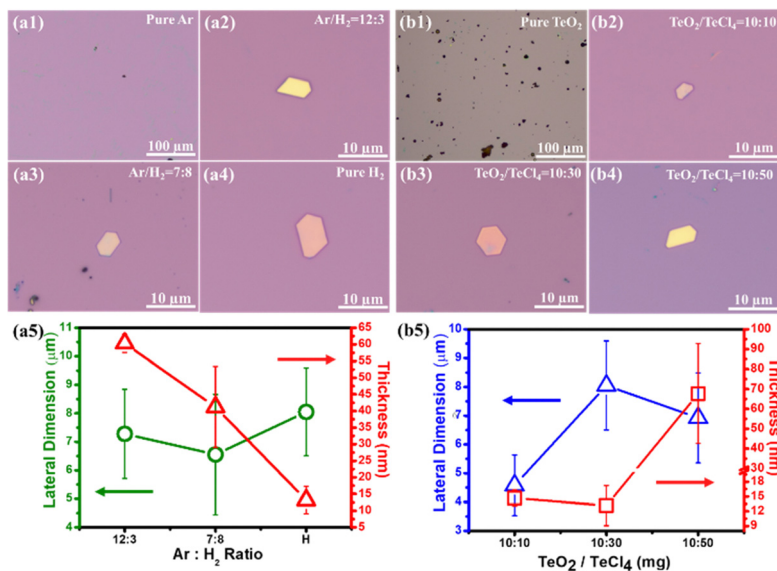


Fig. 3 Investigation of the gas ratio of Ar/H₂ and the ratio of TeO₂/TeCl₄ precursors on the formation of tellurine. Tellurine formed in the presence of (a1) pure Ar 15 sccm, (a2) Ar/H₂ = 12 : 3 (sccm), (a3) Ar/H₂ = 7 : 8 (sccm), and (a4) pure H₂ 15 sccm. (a5) The dependence of the lateral dimension of tellurine on the gas ratio of Ar/H₂, (b1) Pure TeO₂, (b2) TeO₂/TeCl₄ = 10 : 10 (mg), (b3) TeO₂/TeCl₄ = 10 : 30 (mg), and (b4) TeO₂/TeCl₄ = 10 : 50 (mg). (b5) The dependence of the lateral dimension of tellurine on the ratio of TeO₂/TeCl₄.

(Fig. 3(b1)), as confirmed by SEM and Raman examination (Fig. S2(a1 and a2), ESI[†]). It was speculated that since the process temperature (750 °C) was slightly above the melting point of TeO₂ (733 °C), only a limited amount of TeO₂ would have melted, resulting in Te vapor deficiency. Whereas, while solely using TeCl₄ as the precursor, one-dimensional nanorods were realized, as shown in Fig. S2(b1 and b2) (ESI[†]), revealing the tendency of its growth along the Z-axis because of the highly anisotropic bonding strength, in accordance with the previous works on synthesizing Te nanowires.^{29,30} It is expected that the low-melting-point TeCl₄ (230 °C) will be easily evaporated and quickly form a saturated vapor. Meanwhile, when reacted with H₂, the TeCl₄ vapor was reduced to Te vapor. Hence, as the temperature gradually increased, the reduction of TeCl₄ by hydrogen became more intense, resulting in an increase in nucleation probability on the SiO₂/Si substrate. Moreover, the Te vapor preferably tended to accumulate on the existing nucleation sites, leading to one-dimensional nanowire growth.³¹ Herein, the influence of the precursor ratio of TeO₂/TeCl₄ on the growth mechanism was elucidated. At a ratio of TeO₂/TeCl₄ = 10:10 mg, hexagonal flakes were synthesized on the SiO₂/Si substrate perfectly (Fig. 3(b2)), and the lateral size increased with the amount of TeCl₄ (Fig. 3(b3)). With excessive TeCl₄ over 50 mg, the color obviously turned shiny golden (Fig. 3(b4)). Moreover, upon increasing the amount of TeCl₄ from 10 to 30 and 50 mg, obviously the density of tellurine increased from 2 μm⁻² to 50 μm⁻², which can be attributed to the increasing Te vapor, which is beneficial to initial nucleation for subsequent growth (Fig. S3, ESI[†]). Finally, the dependence of the ratio of TeO₂/TeCl₄ on the lateral size and thickness was plotted, as shown in Fig. 3(b5), which displays that the thickness greatly increased from 12 nm to 70 nm while the lateral size did not undergo great

changes (Fig. 3(b5)). The strong tendency toward an increase in thickness with the increasing amount of TeCl₄ yielded a preferential vertical growth, which is in accordance with previous works on the synthesis of one-dimensional nanowires by using TeCl₄ as the Te source.³² Hence, by precisely controlling the ratio of TeO₂/TeCl₄, it is possible to manipulate the growth mechanism of tellurine from one-dimensional nanowire to two-dimensional tellurine. In addition, the process temperature has a great influence on the thickness of tellurine, as shown in Fig. S4 (ESI[†]). It was found that tellurine could not be synthesized when the process temperature was 680 °C. However, when the temperature reached 720 °C, hexagonal Te flakes formed on the SiO₂/Si substrate successfully. When the process temperature was elevated to 750 °C, the lateral size of tellurine increased to 8 μm. However, when the process temperature was elevated to 780 °C, the thickness increased from 12 nm to 90 nm, implying that the excessive Te vapor preferably led to epitaxial growth along the Z-axis. Therefore, the growth window of tellurine is considerably narrow and significantly correlates with the ratio of TeO₂/TeCl₄, the gas mixture of Ar/H₂, and the process temperature.

Moreover, with the aid of TeCl₄, tellurine was not only successfully synthesized on the SiO₂/Si substrate but also could be demonstrated on quartz, mica and sapphire (Fig. S5, ESI[†]).

Apparently, the thickness of tellurine on quartz was found to be much greater than that on other single crystalline substrates owing to the greater roughness, which would lead to the localization of the Te vapor, resulting in the intensive deposition of Te. In addition, the high density of hexagonal flakes spread on mica and sapphire is due to the high crystallinity and low roughness of the underlying substrates, which are beneficial to the epitaxial growth of tellurine.



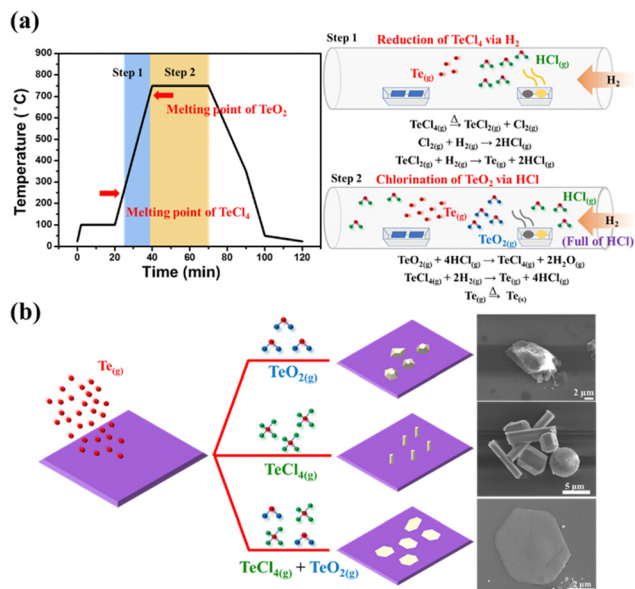
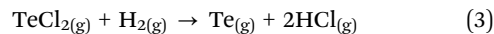
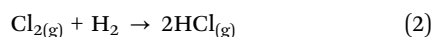
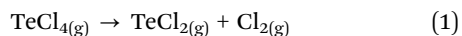


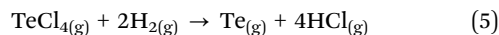
Fig. 4 (a) The temperature profile of the CVD process corresponding to the two-step chemical reaction. Step 1: the reduction of TeCl_4 via H_2 ; step 2: The chlorination of TeO_2 via HCl . (b) The reaction route of the formation of tellurene depending on the selection of precursors.

Finally, the detailed growth mechanism of tellurium from bulk down to 1D and 2D controlled by the precursors $\text{TeCl}_4/\text{TeO}_2$ is well-established. It is worth mentioning that, while solely using TeO_2 as the precursor, it was discovered that not only the products of the Te bulk but silver residues also remained in the boat wherein TeO_2 was initially placed after the process (Fig. S6(a), ESI[†]); material characterization by Raman spectroscopy (Fig. S6(b), ESI[†]) confirmed the existence of $\alpha\text{-TeO}_2$.³³ However, in the case of using the mixture of $\text{TeCl}_4/\text{TeO}_2$ as the Te precursors, no residues were observed (Fig. S6(c), ESI[†]), indicating that TeO_2 was completely and effectively reduced to Te with the assistance of TeCl_4 . The routes of synthesizing tellurene are illustrated in Fig. 4(a). As the process temperature ramps up over the melting point of TeCl_4 (230 °C), TeCl_4 is dissociated into TeCl_2 and Cl_2 ,³⁴ and reacts with H_2 . Eventually, the Te vapor is generated through the reduction reaction by H_2 accompanied by the formation of the massive byproduct HCl .³⁵ When the temperature approaches the melting point of TeO_2 (733 °C), the TeO_2 vapor reacts with HCl to form TeCl_4 ³⁶ and H_2O . These intermediate products successively react with H_2 to generate the Te vapor as a result. It is worth noting that through DFT calculations, it has been proven that the intermediate H_2O benefits the construction of the layered structure of Te.²¹ Such a high concentration of Te vapor mitigates the inert chemical reactivity itself, turning conducive to the formation of nucleation sites on the SiO_2/Si substrate at the early stage. The chemical reaction involved in the deposition of Te can be divided into two steps.

Step 1: reduction of TeCl_4 by H_2



Step 2: chlorination of TeO_2 by HCl



Summarizing the above experimental results, the selection of Te precursors plays a pivotal role in determining the resultant dimension of Te. With only TeO_2 as the Te precursor, in the reduction process, only irregular Te bulk is formed. Moreover, with only TeCl_4 as the Te precursor, one-dimensional nanowire growth is dominant.

Nevertheless, with the addition of low-melting-point TeCl_4 with TeO_2 as Te precursors, hexagonal flakes of tellurene are successfully synthesized on the SiO_2/Si substrate due to the high saturated vapor pressure achieved by the addition of low-melting chloride. Hence, it is speculated that the consistent supply of Te feedstocks is essential to overcome the inherent limitation in growth direction imposed by the high anisotropy in the atomic structure, promoting the lateral growth of the (001) plane and leading to the formation of tellurene. Fig. 4(b) depicts the different dimensions of Te controlled by the different Te precursors. Additionally, based on the previous reports on synthesizing Te nanowires via the hydrothermal approach, we found that the surfactants PVP and CPK can modulate the growth direction of Te NWS, inhibiting lateral growth at an optimized ratio of PVP and CPK. Therefore, it was speculated that the usage of chloride can provide a similar function of altering the growth direction of Te from vertical to lateral growth, resulting in the growth of a two-dimensional film.³⁷

Due to high anisotropy in the bonding strength of two-dimensional materials in which the atoms exist in a covalently bonded plane and the cross planes are attracted by van der Waals force, an intriguing thermal transport property is observed, which has been explored in graphene,³⁸ MoS_2 ,³⁹ and WSe_2 .⁴⁰ Recently, the anisotropic thermal conductivity of tellurene was revealed, exhibiting an in-plane thermal conductivity of $2.5 \text{ W m}^{-1} \text{ K}^{-1}$ and cross-plane thermal conductivity of $1.7 \text{ W m}^{-1} \text{ K}^{-1}$, as measured by micro-Raman thermometry and the time-domain thermal reflectance (TDTR) method.¹⁷

Scanning Thermal Microscopy (SThM) is an emerging technique, which is based on the model of scanning probe microscopy (SPM), with a thermally sensitive probe allowing the microanalysis of thermal properties over a nanoscale area where the tip is scanned. In this study, the active SThM employed and the working mechanism are described as follows: the thermal properties of the material are investigated by connecting a probe to a Wheatstone bridge inside the probe station, which consists of two known resistors denoted as R_1 and R_2 , a variable resistor denoted as R_3 and a thermal probe resistor denoted as R_{Tip} . Once a constant voltage (V_s) is applied to heat up the probe by the Joule heating effect, the heat flow is conducted to the sample through the tips. When the tip sweeps the surface of the sample, the heat flow is transferred from the



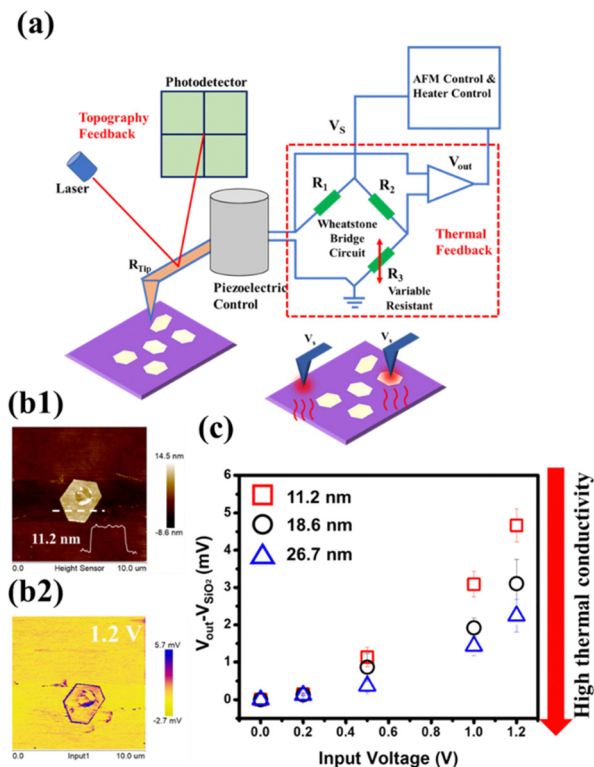


Fig. 5 (a) The schematics of SThM, including the Wheatstone bridge circuit, consisting of two known resistors, R_1 and R_2 , one variant resistor, R_3 , and a scanning tip controlled by the piezoelectric effect. (b1) The AFM images of tellurene flakes with a thickness of 11.2 nm. (b2) The SThM mapping images of tellurene flakes with an applied voltage of 1.2 V. (c) SThM response as a function of the thickness of the tellurene flakes with increasing voltage.

tip to the sample, and the resistance of the probe (R_{Tip}) changes. Meanwhile, in order to maintain the probe temperature, the feedback loop will alter the applied current to the probe during the scanning process, and R_3 will record the resistance change of the probe output (V_{out}), that is the temperature change during the scanning process.⁴¹ Fig. 5(a) depicts the SThM system. Due to the high temporal (1 mK) and lateral spatial resolution (100 nm) provided by SThM,⁴² it has gradually garnered a lot of attention in the exploration of the thermal properties of two-dimensional materials, such as graphene on oxidized Si,⁴³ MoS₂ on the SiO₂/Si substrate,⁴⁴ the heterostructure of MoS₂/WSe₂⁴⁵ and InSe, revealing the anomalous thermal property due to the interfacial phonon transport.⁴⁶ However, there are still challenges regarding precise data reading, temporal response and spatial resolution when the size of the devices decrease.⁴¹ Here, SThM was exploited, for the first time, to reveal the thermal properties of tellurene. Tellurene flakes with different thicknesses, including 11.2 nm, 18.6 nm and 26.7 nm, on the SiO₂/Si substrate were prepared for the examination. Fig. 5(b1) presents the AFM images of a representative tellurene flake, and the thickness was confirmed to be 11.2 nm. Fig. 5(b2) displays the mapping of the SThM response (applied voltage 1.2 V), which matches well with the appearance presented by AFM. In addition, in the SThM image, the tellurene flakes displayed a

darker yellow with respect to the yellow background of the SiO₂ region, indicating that tellurene had higher thermal resistance than that of SiO₂. Some areas in the centre presented a purple color due to surface defects or protrusions, affecting the phonon transport, resulting in poor thermal conductance.⁴⁷ According to the previous report regarding the thermal behavior of graphene,⁴⁷ the thermal transport is greatly affected by the surface roughness and protrusions, and resulting in the suppression of phonon transport. Hence, the SThM technique is able to provide more precise and local analysis, such as edges and defects in a tiny area, which is not possible with the conventional Raman examination.^{48–50} Once the tip scans from the background (SiO₂) to tellurene at a specific applied voltage, the different thermal conductance of tellurene and SiO₂ result in a change in the resistance or voltage of the tip, and then to maintain the initial status of the tip, additional voltage is exerted, which is denoted as V_{out} . V_{SiO_2} was set as the reference voltage. Hence, a larger value of $V_{\text{out}} - V_{\text{SiO}_2}$ stands for lower thermal conductance. It was noted that the signal from the edge of the flakes was varied probably due to the limited contact area and the unidirectional heat flow.^{46,51}

Next, to explore the influence of flake thickness on the thermal property, tellurene flakes with different thicknesses, including 11.2 nm, 18.6 nm and 26.7 nm, were prepared for the examination. The results are shown in Fig. 5(c). When the applied voltage was increased, the value of $V_{\text{out}} - V_{\text{SiO}_2}$ increased regardless of the thickness. Moreover, at the same applied voltage, the 11.2 nm tellurene had a higher value of $V_{\text{out}} - V_{\text{SiO}_2}$ than that of tellurene with higher thickness, indicating a decrease in lateral thermal conductance due to the increase in phonon-boundary scattering.^{52,53} For tellurene with thicknesses 18.6 and 26.7 nm, the thermal conductance increased because the behavior was approaching the metallic property.¹⁷

To examine the electrical transport properties of the synthesized tellurene samples, field effect transistors (FET) were fabricated on the SiO₂/Si substrate, in which the Si substrate was taken as the back gate to modulate the electrostatic potential of the tellurene device, similar to a field-effect transistor. Through electron-beam lithography followed by thermal evaporation, the drain-source contacts were formed with Ti/Au (20/80 nm) electrodes. A schematic of the tellurene device with the circuit layout is shown in Fig. 6(a). The inset shows the typical optical image of the multi-electrode tellurene device consisting of the Ti/Au electrodes on tellurene on the top surface of the SiO₂/Si substrate. In this work, several Te devices were fabricated to show highly consistent results, as follows. The output characteristics ($I_{\text{ds}} - V_{\text{ds}}$) of the Te device, expressed at different V_{bg} and room temperature are displayed in Fig. 6(b). All the I_{ds} values increased linearly with V_{ds} , suggesting the formation of an Ohmic contact between tellurene and the Ti/Au electrodes. The corresponding transfer characteristics ($I_{\text{ds}} - V_{\text{bg}}$) of the Te device are given in the inset. No obvious gate-voltage modulation could be observed, thus further temperature-dependent electrical measurements were needed to explore charge transport in the synthesized tellurene. The transfer characteristics of the tellurene device at different temperatures



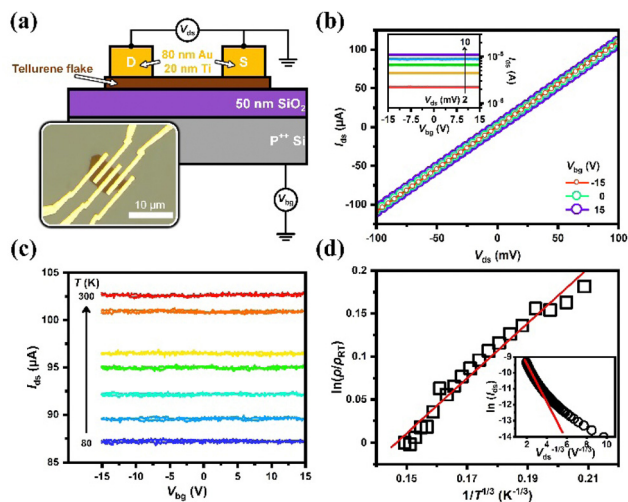


Fig. 6 (a) The schematic of a tellurene device used in this work, with the circuit diagram overlaid. The heavily doped Si substrate was used as the back gate. The pair of electrodes served as the source (S) and the drain (D). The inset shows a typical optical image of the tellurene device. (b) The output characteristics of the tellurene device at room temperature under different V_{bg} values. The inset shows the corresponding transfer characteristics at different V_{ds} from 2 to 10 mV. (c) The transfer characteristics of the tellurene device at different temperatures. (d) Normalized resistivity as a function of temperature for the tellurene device (room-temperature values). The solid line presents the best fit for the data. The inset shows the I_{ds} - V_{ds} relationship of the Te device in the form of $\ln(I_{ds}) - V_{ds}^{-1/3}$. The solid line shows the high-field approaching behavior.

from 300 down to 80 K are shown in Fig. 6(c). The I_{ds} values were observed to decrease with decreasing temperature, rather than metallic transport features, which implies the thermal activation of a semiconductor or a disordered hopping system. Fig. 6(d) further presents the normalized resistivity of the tellurene device in a logarithmic scale, showing its linear dependence on $T^{-1/3}$ over the entire measured temperature range, which indicates that the low-temperature transport is governed by the model of 2D variable-range hopping (VRH). The solid line in red shows the best fit to the data according to the equation $\rho(T) = \rho_{RT} \exp[(T_0/T)^{1/3}]$.⁵⁴ All the Te devices studied in this work had room-temperature resistivity ranging from 50–80 $\Omega \mu\text{m}$ and showed the same temperature-dependent behavior. To verify that the transport in tellurene follows the 2D VRH model, the output curve under high electric fields at 110 K is shown in the form of a plot⁵⁵ in the inset of Fig. 6(d), which supports the presence of the 2D VRH model in tellurene, as well as the 2D configuration for the carriers.

Conclusions

In summary, tellurene synthesized on the SiO_2/Si substrate *via* a CVD process is demonstrated for the first time in this work. The Raman spectra of the hexagonal flakes of tellurene display a thickness-dependent evolution of the A1 peaks, which red-shifted with increasing thickness.

Moreover, the growth parameters, such as the gas mixture ratio and the ratio of $\text{TeCl}_4/\text{TeO}_2$, were systematically

investigated, confirming the role of the precursors $\text{TeCl}_4/\text{TeO}_2$ in the formation of tellurene, and finally, the growth mechanism is established. The fresh tellurene flakes exposed to the air exhibit remarkable stability over 30 days, as confirmed by TEM observation, which is a significant step forward to realizing more practical electronic device applications. Moreover, the thermal property of tellurene was well-investigated *via* an emerging technique SThM, enabling the imaging of lateral thermal transport and finally proving that the thermal conductivity is proportional to the thickness of tellurene due to the contribution of cross-plane phonon transport. Finally, bottom-gated FETs based on tellurene flakes were fabricated to evaluate the electrical properties. From the output characteristics, the I_{ds} values decreased with decreasing temperature, which is the feature of semiconductor transport and implies the thermal activation of a semiconductor or a disordered hopping system. By interpreting the normalized resistivity of the Te device in the logarithmic scale, its linear dependence on $T^{-1/3}$ in the entire temperature range was evident, which matches well with the model of 2D variable-range hopping (VRH) for charge transport.

Experimental section

Synthesis of tellurene flakes by the chemical vapor deposition

The precursor powders of (10 g) TeO_2 and (30 g) TeCl_4 were separately placed in a quartz boat upstream of the furnace, and the SiO_2/Si substrate was placed 30 cm away from the quartz boat. Next, a large amount of pure Ar was steadily introduced for 20 minutes to remove the impurities in the tube, and the pressure was controlled to 700 torr. Then, 15 sccm H_2 was injected into the quartz tube, and the temperature of $\text{TeO}_2/\text{TeCl}_4$ and the SiO_2/Si substrate was maintained at 750 °C respectively for 30 minutes for the deposition of tellurene. After the process, pure Ar was introduced, and the system was slowly cooled to room temperature.

Characterization

The crystallinity of the tellurene flakes was examined by a Raman spectrometer (HORIBA Jobin-Yvon, iHR550) equipped with green laser (532 nm) excitation. Prior to acquiring the atomic image, the sample was prepared using a focused ion beam (FIB, FEI Helios G3CX). The atomic structure of tellurene was examined by transmission electron microscopy (TEM, JEOL JEM-F200) at an image resolution of 0.19 nm using a high-resolution energy-dispersive spectrometer (EDS). The thickness and dimension of tellurene were acquired by atomic force microscopy (AFM, Bruker Dimension Icon). The microstructure of tellurene was revealed by scanning electron microscopy (SEM, HITACHI SU-5000). The thermal images were measured by using an SThM probe (VITA-DM-GLA) mounted on the tip cantilever of a Bruker Dimension Icon AFM. Typically, 512×512 -pixel scans were performed at a line rate of 0.2 Hz.



Author contributions

Y. H. C. and Y. C. C. conducted this work and data analysis. Y. H. C., Y. C. C. and Y. Z. C. conceived and coordinated the research. Y. H. C., Y. C. C. drew the curves and figures. C. Y. L. fabricated the electrical devices and measured the electrical properties. Alice C. C. conducted SThM. Y. F. L. and Y. Z. C. provided theoretical guidance. All authors discussed the results and commented on this manuscript. Y. Z. C. wrote the manuscript with contributions from all co-authors.

Conflicts of interest

There are no conflicts to declare.

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