



Cite this: *Phys. Chem. Chem. Phys.*,
2023, 25, 30785

Received 29th August 2023,
Accepted 23rd October 2023

DOI: 10.1039/d3cp04170k

rsc.li/pccp

Recent advances in copper chalcogenides for CO₂ electroreduction

Wenjian Hu, ^{ab} Didier Grandjean, ^b Jan Vaes, ^{ac} Deepak Pant ^{*ad} and Ewald Janssens ^{*b}

Transforming CO₂ through electrochemical methods into useful chemicals and energy sources may contribute to solutions for global energy and ecological challenges. Copper chalcogenides exhibit unique properties that make them potential catalysts for CO₂ electroreduction. In this review, we provide an overview and comment on the latest advances made in the synthesis, characterization, and performance of copper chalcogenide materials for CO₂ electroreduction, focusing on the work of the last five years. Strategies to boost their performance can be classified in three groups: (1) structural and compositional tuning, (2) leveraging on heterostructures and hybrid materials, and (3) optimizing size and morphology. Despite overall progress, concerns about selectivity and stability persist and require further investigation. This review outlines future directions for developing the next-generation of copper chalcogenide materials, emphasizing on rational design and advanced characterization techniques for efficient and selective CO₂ electroreduction.

1. Introduction

The worldwide need for energy continues to grow while the alarming rise in greenhouse gas (GHG) emissions demand for sustainable and eco-friendly solutions.^{1–8} Carbon dioxide (CO₂) is a major GHG, contributing significantly to global warming and shifts in climate patterns. Creating practical and affordable solutions to capture and transform CO₂ into valuable chemicals and fuels is thus of urgent importance.^{9–11} Among various CO₂ conversion methods that are being developed, electrochemical reduction has emerged as one of the most appealing approaches, given its notable energy efficiency, its potential for renewable energy integration, and its capacity to generate a variety of valuable products.^{12–27} Specifically, the process of CO₂ electroreduction using Cu-based catalysts^{22,27–44} has garnered tremendous attention because Cu demonstrates notable activity for C–C coupling, essential to the formation of multicarbon products.

Copper chalcogenides are a family of compounds that combine copper with elements from the chalcogen group, comprising sulfur, selenium, and tellurium. These materials have attracted considerable interest for multiple uses in semiconductors,

photoconductors, thermoelectric materials, and solar cells.^{45–50} Copper chalcogenide bonds are highly covalent, although the chalcogenide is treated as a divalent anion (S^{2–}, Se^{2–}, or Te^{2–}) in some cases. In a basic lattice cell, chalcogen atoms constitute the frame of a regular lattice, whereas the copper atoms are disordered, moving like a liquid and taking random sites in the chalcogen lattice.^{51,52} This phenomenon leads to variations in stoichiometry, which is represented by the characteristic formula Cu_{2–x}E (E = S, Se, Te) with possible x values ranging from 0 to 1.^{53,54} The copper chalcogenide family encompasses various binary, ternary, and quaternary compounds, whose stoichiometries determine the crystal structure. The binary copper sulfide system has monoclinic (Cu₃₁S₁₆ and Cu_{1.94–1.97}S), cubic (Cu₉S₅ and Cu_{1.8}S), orthorhombic (Cu₇S₄ and Cu_{1.75}S), triclinic (Cu₅₈S₃₂ and Cu_{1.81}S), and hexagonal (CuS) crystal phases. The binary copper selenide system has cubic (Cu₂Se and Cu_{1.8}Se), tetragonal (Cu₃Se₂), hexagonal (CuSe and Cu_{0.87}Se), and orthorhombic (Cu₅Se₄ and CuSe₂) phases. Known crystal phases of copper tellurides are hexagonal (Cu₂Te), tetragonal (Cu₇Te₄, Cu₄Te₃ and Cu₃Te₂), orthorhombic (CuTe), and pyrite-type (CuTe₂).^{55–57} In the context of the electrochemical CO₂ reduction, copper chalcogenides have shown an exceptional selectivity as a result of their distinct electronic and structural properties.

The synergistic effect between copper and the chalcogen elements is responsible for the remarkable electrocatalytic CO₂ reduction performance of these compounds. The presence of chalcogen atoms modulates their electronic structure, in particular the location of the Cu d-band center altering the adsorption behavior of the reaction intermediates, and subsequently influencing the catalyst's specificity and effectiveness.^{54,58} The tunability

^a Separation and Conversion Technology, Flemish Institute for Technological Research (VITO), Boeretang 200, 2400 Mol, Belgium. E-mail: deepak.pant@vito.be

^b Quantum Solid-State Physics, Department of Physics and Astronomy, KU Leuven, Celestijnenlaan 200 D, 3001 Leuven, Belgium.

E-mail: ewald.janssens@kuleuven.be

^c Department of Solid-state Sciences, Ghent University, Krijgslaan 281/S1, 9000 Gent, Belgium

^d Center for Advanced Process Technology for Urban Resource Recovery (CAPTURE), Frieda Saeysstraat 1, 9052 Zwijnaarde, Belgium



of the copper chalcogenide properties enables the optimization of their electrocatalytic performance through various strategies, such as controlling their composition, adjusting their crystal structure and morphology, and creating heterostructures or hybrid materials. These approaches can be employed to customize the electronic configuration and catalytic performance of copper chalcogenides, enhancing their performance in CO_2 electroreduction by achieving higher selectivity, improved stability, and increased energy efficiency.⁵⁹

The objective of this review is to examine the latest developments in copper chalcogenide catalysts for CO_2 electroreduction, focusing on the most significant contributions from the past five years. We discuss the latest approaches to enhance their efficacy as electrocatalysts, including structural and compositional tuning, exploiting heterostructures and hybrid materials, and optimizing their shape and particle size by synthesis methods.

Although significant advancements have been made in recent times, obstacles persist in increasing the preference for particular products and boosting the enduring stability of copper chalcogenide catalysts. Therefore, we also outline future perspectives on the development of next-generation copper chalcogenide materials, emphasizing the importance of rational design strategies and the integration of advanced characterization techniques to achieve efficient and selective CO_2 electroreduction.

2. Synthesis strategies for copper chalcogenides

Copper chalcogenides have been studied for several decades, initially for their applications in superconductors and thermoelectric materials. Their synthesis has evolved over time with

enhanced control over their composition, structure, and morphology. Solvothermal methods, vital for nanoparticle synthesis, allow the production of highly crystalline materials under controlled temperatures and pressures, with the ability to tailor compositions and structures. Electrosynthesis, offers fine control over the material properties and allows for *in situ* formation of compounds like copper chalcogenides on conductive substrates, but it does present challenges in scaling up. Meanwhile, cation exchange methods permit careful post-synthesis modifications, control over composition, and the synthesis of complex heterostructures and hybrid materials. By combining these techniques, researchers could overcome their individual limitations; solvothermal synthesis forms the initial nanostructures, electrosynthesis deposits additional material, and cation exchange refines their final properties, broadening the horizons for nanomaterial design and innovation (Fig. 1). Other methods, such as wet chemical method^{60–62} pyrolysis⁶³ and heat treatment^{64,65} have also been explored to produce copper chalcogenide materials with various catalytic properties for the electrochemical CO_2 reduction reaction (e CO_2 RR). Inspired by the water evaporation technique used in ref. 66, we could synthesize a variety of Cu_{2-x}Se nanostructures, including nanosheets, nanoflowers, and nanowires. This was achieved by variations of the Cu/Se ratio of 3.75/19, 5/19 and 7.5/19, respectively (Fig. 2).

2.1 Solvothermal synthesis

Solvothermal synthesis is a prominent method for preparing copper chalcogenides for e CO_2 RR. It uses elevated temperatures and pressures in a sealed reactor to produce highly crystalline materials with tailored properties. By adjusting the

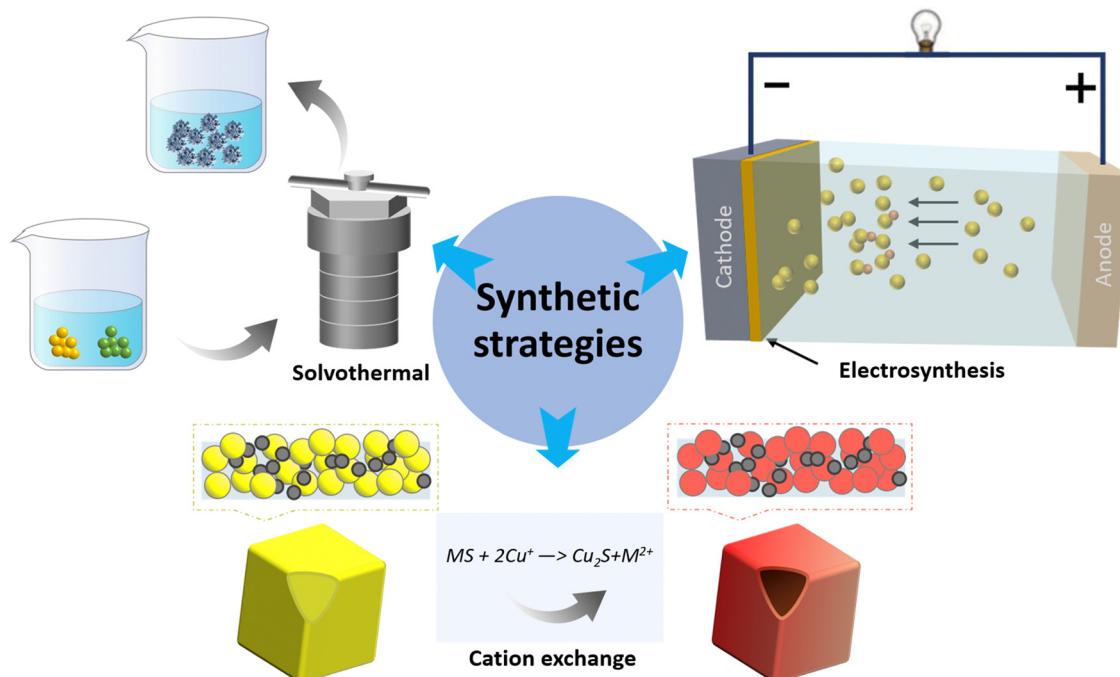


Fig. 1 Schematic diagram of synthesis strategies, including solvothermal, electrosynthesis and cation exchange. M stands for metal.



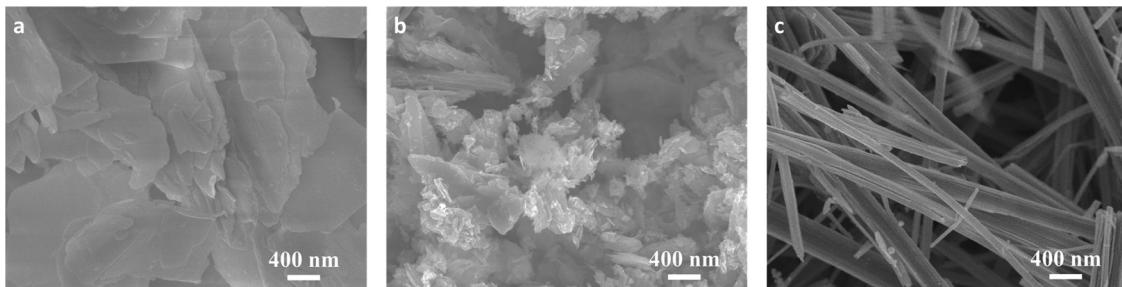


Fig. 2 Scanning electron microscopy images illustrating the variety of Cu_{2-x}Se nanostructure that can be grown by changing the Cu/Se ratio. (a) Nanosheets, (b) nanoflowers, and (c) nanowires synthesized from a Cu/Se precursor ratio of 3.75/19, 5/19 and 7.5/19, respectively.

reaction parameters, various nanostructures, such as nanoparticles, nanowires, and nanosheets could be obtained.

Solvothermal methods have extensively been used to synthesize copper sulfide nanostructures.^{63,67-82} For example, Shinagawa *et al.*⁶⁷ reported that CuS nanocatalysts can be produced by *in situ* reductive reformation of CuS precursor materials. They developed a new method to create carbon-supported copper sulfide nanoparticles by mixing copper nitrate, thiourea, urea, and carbon black in water, and heating the mixture to precipitate the nanoparticles. Wang *et al.*⁷³ utilized a combination of solvothermal and electrochemical techniques to create Cu_2SnS_3 nanosheets. This involved dissolving $\text{SnCl}_4 \cdot 5\text{H}_2\text{O}$ and $\text{Cu}(\text{CH}_3\text{COO})_2 \cdot \text{H}_2\text{O}$ within a combined concoction of ethylene glycol and water, agitating the solution, adding thioacetamide, and heating the mixture in a Teflon-lined autoclave. The synthesis of CuS nanosheets followed a similar procedure, while SnS_2 nanosheets were prepared using $\text{SnCl}_4 \cdot 5\text{H}_2\text{O}$, sodium dodecyl benzenesulfonate, and L-cysteine. Following eCO₂RR, the samples experienced self-adapted phase separation, resulting in the formation of $\text{SnO}_2 @ \text{CuS}$ and $\text{SnO}_2 @ \text{Cu}_2\text{O}$ heterojunctions. Zhang *et al.*⁶⁹ created CuS nanosheets by mixing 6 mL of 0.5 M thioacetamide with 25 mL of 0.2 M $\text{Cu}(\text{NO}_3)_2$ ethanol solutions. The mixture was heated to 90 °C for 3 hours, whereafter the resulting black precipitate was collected. CuS hollow particles were synthesized using a modified method in which thioacetamide dissolved in ethanol is added to a dispersion of MOF-199 in ethanol that is subsequently heated at 90 °C for 1 hour. TEM and high-angle annular dark-field scanning TEM (HAADF-STEM) images showed that CuS hollow particles

have an average size of 1.3 μm and are composed of CuS nanosheets with a thickness of about 3 nm.

The hydrothermal synthesis was also used to produce copper selenide nanomaterials.⁸³⁻⁸⁹ Yang *et al.*⁸³ combined $\text{CuCl}_2 \cdot 2\text{H}_2\text{O}$, Na_2SeO_3 , and hydroxylamine in a diethylenetriamine and deionized water mixture with varying volume ratios to synthesize $\text{Cu}_{2-x}\text{Se}(\text{y})$ nanocatalysts. After stirring for 30 minutes, the mixture was warmed in a Teflon-coated pressure vessel at 180 °C during 15 hours and the resulting products were washed, and dried. The obtained $\text{Cu}_{1.63}\text{Se}$ nanoparticles had a typical size of 50 nm (see Fig. 3). Wang *et al.*⁸⁵ prepared CuInSe_2 using a modified solvothermal method. Oleylamine was degassed and purged with N_2 before combining CuCl , InCl_3 and Se in a flask. The mixture was heated and stirred before the products were washed, and dried. Duan *et al.*⁸⁸ synthesized Cu_2Se nanosheets using a hydrothermal treatment. BiCuSeO nanosheets were produced by mixing $\text{Bi}(\text{NO}_3)_2 \cdot 5\text{H}_2\text{O}$ with $\text{Cu}(\text{NO}_3)_2 \cdot 3\text{H}_2\text{O}$, KOH and NaOH, heating the mixture in a Teflon-lined vessel, and washing and drying the obtained precipitate. The average thickness of the BiCuSeO sheets was 9.2 nm. High-resolution TEM images display two perpendicular sets of lattice fringes with a spacing of 0.28 nm, verifying the successful synthesis of ultrathin BiCuSeO single-crystal nanosheets. Ding *et al.*⁸⁷ synthesized K-doped Cu_2Se nanosheet arrays on a Cu foam. The foam was cleaned, soaked in a mixture of Se powder, NaBH_4 , NaOH, deionized water, and KBr, and heated in an autoclave. Various KBr amounts were used to produce different potassium dopant contents, while undoped pure Cu_2Se was obtained without KBr. 11.2% K-doped Cu_2Se had

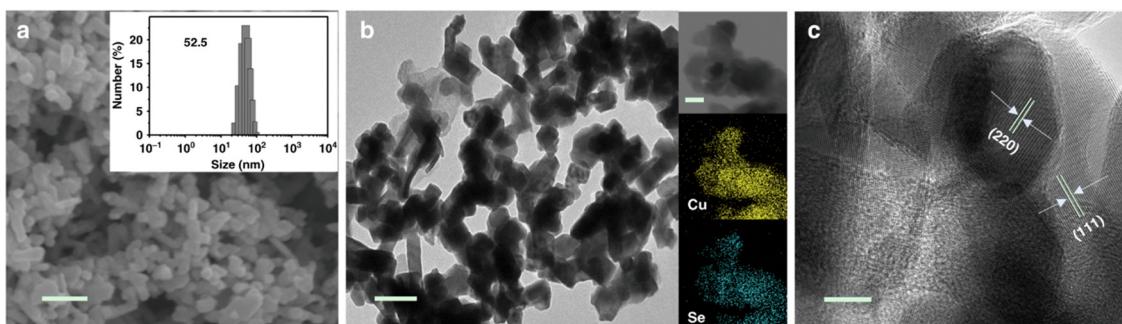


Fig. 3 Structural characterization of $\text{Cu}_{1.63}\text{Se}$ nanocatalysts. (a) SEM image with a 200 nm scale bar. The inset presents the size distribution as obtained by dynamic light scattering (DLS) measurements. (b) TEM image with a 100 nm scale bar, including elemental mappings as insets. (c) HRTEM image with a 10 nm scale bar. Copyright 2019 Nature Publishing Group.⁸³



a honeycomb-like structure and a thickness of 1.4 μm , while pure Cu₂Se had a thickness of 1.8 μm . Those nanosheets display a plane spacing of 0.205 nm, which correlates with the (220) facet of cubic Cu₂Se.

The latter method is widely used for the synthesis of various types of materials including nanoparticles, single crystals, and complex structures. It provides the advantage of synthesizing high-quality materials at relatively lower temperatures than conventional methods. However, the process is often time-consuming and requires careful handling due to high pressures involved. In addition, the solvents used can be harmful, requiring careful management.

2.2 Electrosynthesis

In situ electrochemical synthesis, including electrochemical deposition, recently gained traction for the synthesis of copper chalcogenide nanomaterial.^{75,81,90–94} Redox reactions cause selective ion deposition on either the cathode or the anode surface. Key benefits of electrochemical deposition include simplicity of the operating procedure, precise control over particle size, and environment friendly outcomes without corresponding oxidation or reduction byproducts. As the understanding of catalyst synthesis methods evolves, many catalysts are not produced using a single method but by combining solvothermal and *in situ* electrochemical synthesis techniques.

Studies emphasize the importance of the catalysts' elemental composition and their intricate microstructures. Cheng *et al.*⁸¹ demonstrated that introducing surface nitrogen can notably improve the electrocatalyst's performance by raising the concentration of active sites and optimizing their electronic structure. The obtained N-enriched Sn/SnS nanosheets demonstrated a five-fold rise in current density and a 2.45-fold increase in faradaic efficiency (FE) compared to their unmodified SnS₂-derived counterparts. The formate production rate was 14 times higher in the N-enriched sheets than in the unmodified ones, reaching 1358 $\mu\text{mol h}^{-1} \text{cm}^{-2}$. The surface nitrogen-injection technique was also effectively employed with other precursors such as CuS and In₂S₃, demonstrating its potential as a broader approach. Peng *et al.*⁷⁵ synthesized double sulfur vacancy-rich CuS_x nanosheets for the electro-conversion of CO₂ into C₃ species. In a first stage CuS nanosheets were prepared by dissolving CuCl₂·2H₂O and thioacetamide in separate solutions that were then combined and subsequently heated and washed. Sulfur vacancies were then introduced in the CuS powder by mixing it with polyvinylidene fluoride and coating it on a copper foil that was used as a cathode in a lithium-ion battery. A 0.044 mA cm⁻² current density was applied within a 0.01–3 V *vs.* Li⁺/Li voltage range. CuS_x catalysts with various sulfur vacancy densities were obtained after 1, 10, and 100 charge–discharge cycles. HAADF-STEM images revealed that the CuS_x-10-cycle nanosheets display a hexagonal phase where the 0.283 nm spacing corresponds to CuS (103) facets. The (103) planes were also seen on CuS_x-1-cycle and CuS_x-100-cycle edges. Deng *et al.*⁹⁰ developed a method for the electrochemical deposition of CuS_x films on polished Cu disks by utilizing square wave voltammetry.

The synthesis procedure involved 2000 cycles with a plating solution containing a mixture of CuSO₄·5H₂O, thiourea, and HCl. Active CuS_x and desulfurized CuS_x were prepared for electrochemical CO₂ reduction to investigate the role of sulfur. Phillips *et al.*⁹¹ prepared sulfide-derived-Cu working electrodes *via* pulsed electrodeposition. The active sulfide-derived-Cu layer was formed *in situ* during the first ~15 minutes of electrolysis. Guo *et al.*⁹³ synthesized Cu₂S hollow nanocubes with a significant presence of sulfur vacancies by electrochemical reduction (see Fig. 4). Cu₂S_{1-x} hollow nanocubes were obtained by electro-reducing Cu₂S hollow nanocubes loaded on a gas diffusion electrode (GDE) at −0.3 V *vs.* RHE for 30 minutes. Li *et al.*⁹⁴ prepared Ag and S co-doped Cu₂O/Cu (Ag, S-Cu₂O/Cu) using an electrochemical reduction method at room temperature in a conventional H-cell. Ag–Cu₂S was first prepared through a cation exchange method and used as the working electrode. Electrochemical reduction was then carried out by applying a potential of −1.6 V *versus* a saturated calomel electrode for 30 minutes, while bubbling CO₂ and stirring continuously. Cu₂O/Cu, S-Cu₂O/Cu, and Ag–Cu₂O/Cu were fabricated using the same procedure, with Cu₂O, Cu₂S, and Ag–Cu₂O as working electrodes, respectively.

Electrosynthesis can be more environment friendly than conventional methods since in many cases the use of electricity does not require potentially harmful chemical reagents. On the downside, it can require extensive optimization of individual processes and, for certain reactions, achieving a high selectivity can be challenging.

2.3 Cation exchange

Cation exchange, a wet chemical method,^{61,62,95–97} is a relatively new approach for synthesizing copper sulfide nanocatalysts for eCO₂RR. By selectively replacing the cations of a host material with copper ions, CuS nanosheets and nanoparticles with controlled properties can be produced.

He *et al.*⁹⁶ introduced electrochemically-driven cation exchange (ED-CE), which enables converting a prearranged CoS₂ template into Cu₂S through a three-step process using a standard three-electrode system. A metal–organic framework (MOF) consisting of a Co-containing zeolitic imidazolate framework (Co-ZIF-L) grown on a carbon cloth and Cu foil is electrochemically converted into cobalt sulfides and finally cations are exchanged. In this configuration, the cobalt sulfide film functioned as working electrode and cation exchange template, while a platinum foil and a silver/silver chloride electrode saturated with potassium chloride served as counter and reference electrodes, respectively. To replace the cobalt cations of CoS₂ with copper ions, a 40 mL dimethylformamide solution containing 3 mM copper nitrate trihydrate and 0.1 M lithium perchlorate was used as electrolyte. Chronoamperometry was employed to drive the cation exchange processes, which was performed first at −0.2 V (*vs.* Ag/AgCl) and thereafter at −0.6 V (*vs.* Ag/AgCl) until a targeted value of electrical charge per electrode area was reached. The synthesis procedure is schematically depicted in Fig. 5. This method ensures that the synthesized Cu₂S catalyst keeps the same structure as the CoS₂ template, while maintaining a high density of grain



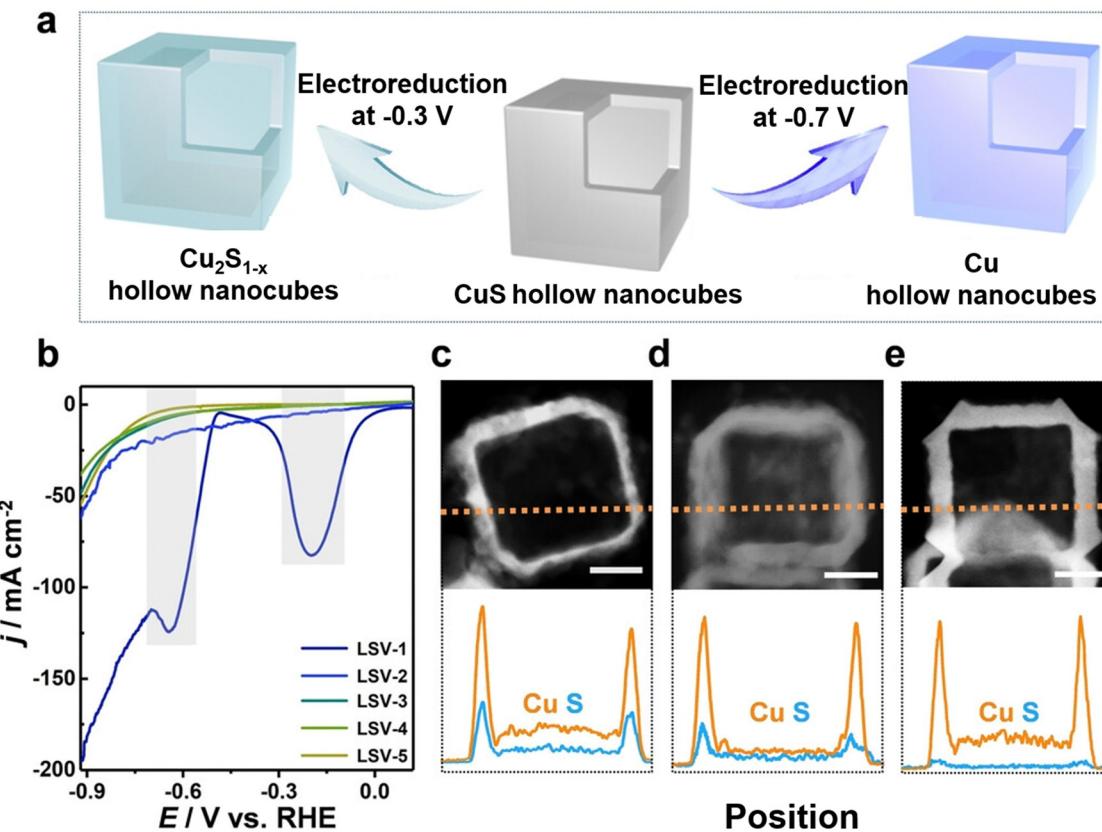


Fig. 4 (a) Schematic representation of the electrochemical reduction process of CuS hollow nanocubes into their $\text{Cu}_{2-\text{x}}$ and Cu counterparts. (b) Subsequent LSV curves for the CuS hollow nanocubes (numbered LSV-1 to LSV-5). TEM images along with corresponding EDX line-scanning profiles for (c) CuS, (d) $\text{Cu}_{2-\text{x}}$, and (e) Cu hollow nanocubes, all featuring a 30 nm scale bar. Figure adapted from ref. 93. Copyright 2022 Wiley VCH.

boundaries. Li *et al.*⁹⁷ developed a simple cation exchange technique to create Ag/Cu electrocatalysts using copper sulfide nanosheets as growth template. The crystal structure of $\text{Cu}_{2-\text{x}}$ S nanosheets with lateral dimensions of 100 nm and a 14 nm thickness gradually transformed from Cu_7S_4 to Ag_2S , with an Ag/Cu mass ratio varying between 0.3 and 25 upon increasing the Ag^+ concentration in the exchange solution. Following the cation exchange process, the nanosheets retain their original morphology, but exhibit shape distortion with increasing Ag content.

We conclude that cation exchange is an attractive method to modify the chemical composition and hence the properties of the material without disrupting its overall structure. It has potential to fine-tune materials for specific applications. However, the method's success highly depends on the cation exchange kinetics and thermodynamics, which can be complex and hard to control.

3. Recent strategies to enhance the performance of copper sulfide in eCO_2RR

Copper sulfide has emerged as an effective catalyst for the eCO_2RR , but further improvement is needed. Notably, the

hydrogen evolution reaction (HER) is a competitive reaction to eCO_2RR that hinders its efficiency. Research has shown a linear correlation between the binding energies of COOH^* / HCOO^* and H^* on most eCO_2RR catalysts, negatively impacting the selectivity.^{98–102} Breaking this correlation is imperative for improving copper sulfide's effectiveness as an eCO_2RR catalyst for industrial applications. Recent strategies to enhance the selectivity and activity of CuS include (i) structural and compositional tuning through sulfur vacancy engineering and foreign metal ions doping, (ii) combining CuS with other materials in heterostructures or hybrid materials to introduce new active sites and improve the catalytic performance through synergistic effects, and (iii) size and morphology control, for example to create nanosheets or hollow structures with optimized surface area and enhanced accessibility of the active sites. These approaches offer a guide for developing effective CuS catalysts for eCO_2RR , advancing eco-friendly conversion methods. A summary of cutting-edge Cu sulfide catalysts, including information on their preparation method, operating conditions and product preferences, is provided in Table 1.

3.1 Structural and compositional tuning

To enhance copper sulfide's eCO_2RR catalytic performance, various structural and compositional tuning

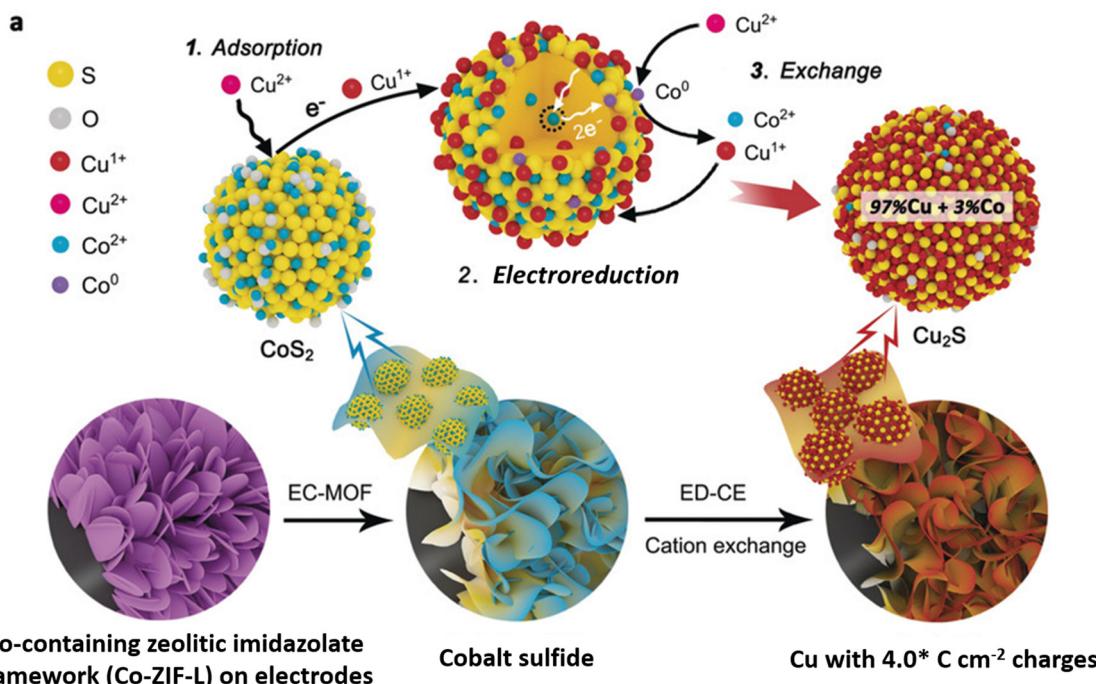


Fig. 5 Experimental routes and the mechanisms involved in cation exchange driven by electrochemical means. Figure adapted from ref. 96. Copyright 2020 Wiley-VCH.

strategies,^{51,65,67,69,75,79,81,82,90,91,93,94,96,97,103} can be employed to adjust its electronic structure and morphology.

We outline two key strategies. The first one is by creating sulfur vacancies in the material during or after its synthesis by controlling the preparation conditions or through post-synthesis treatments, respectively. Their presence increases the likelihood of CO^* formation on the catalyst surface. The second is by doping CuS with metal cations. Ag doping, for instance, induces synergistic interactions between Ag_2S and CuS that suppress HER and improve the selectivity towards CO_2 reduction. Metal doping can be achieved by co-precipitation, ion exchange, or electrochemical deposition.

Cheng *et al.*⁸¹ found that surface nitrogen-injection can increase the number of active sites and optimize the material's electronic structure. N-Enriched Sn/SnS nanosheets demonstrated significant improvements in current density and FE compared to their pristine versions. The improved catalytic activity was confirmed through density functional theory (DFT) calculations to be a result of the alteration in the valence of the Sn sites. The approach is applicable to other metal sulfide precursors as well. Peng *et al.*⁷⁵ showed that double sulfur vacancies (DSVs) in the CuS(100) planes could serve as active electrocatalytic centers for CO_2RR by favoring the creation of n -propanol's key $^*\text{C}_3$ intermediate. They applied a lithium electrochemical tuning method, using the CuS as the cathode and a lithium foil as the anode, yielding promising results for the controlled creation of ion vacancies (see Fig. 6). DFT calculations suggest that the creation processes of $^*\text{HCOOH}$ and CO^* are in competition with the presence of sulfur vacancies, increasing the formation probability of the latter one. During continuous electrochemical CO_2RR testing, the CuS_x -DSV catalyst maintained a

$\text{FE}_{n\text{-PrOH}}$ of 12.1% for 10 hours at -1.05 V vs. RHE, corresponding to a retention of approximately 78%. Meanwhile, an increase in the hydrogen production activity was observed, indicated by a continuous rise in the current. Deng *et al.*⁹⁰ found that sulfur plays a crucial role in this process, with active CuS_x exhibiting high selectivity and activity towards formate ($\text{FE HCOO}^- = 75\%$ and $j \text{ HCOO} = -9.0 \text{ mA cm}^{-2}$ at -0.9 V). Operando Raman spectroscopy revealed that sulfur dopants suppress the formation of CO intermediate during e CO_2RR , resulting in a decreased FE for CO, hydrocarbons, and alcohols. Supporting DFT calculations indicated that the presence of sulfur on the copper surface modified the adsorption strengths of HCOO^* , promoting formate production while suppressing that of CO. Phillips *et al.*⁹¹ suggested that the selective formation of formate is due to the stronger binding of the CO intermediate originating from the sulfur atoms. Guo *et al.*⁹³ reported a $\text{Cu}_2\text{S}_{1-x}$ catalyst, featuring numerous $\text{Cu}^{\delta+}$ species that can produce ethanol at ultralow overpotential with an outstanding selectivity and FE with a 6-hour stability. The integrated findings from *in situ* attenuated total reflection Fourier transformed infrared spectroscopy and DFT calculations demonstrate that the robust electron-donating ability of $\text{Cu}^{\delta+}$ sites contributes to reducing the Gibbs free energy for the C-C coupling step involved in generating ethanol, while concurrently enhancing that associated with formate formation.

Li *et al.*⁹⁴ found that the synergistic interactions between S^{2-} anions and Ag^+ cations dopants and the $\text{Cu}_2\text{O}/\text{Cu}$ host material, not only enhanced the Ag, S- $\text{Cu}_2\text{O}/\text{Cu}$ electrocatalyst selectivity and current density for methanol formation but also ensured its stability over 24 hours. More specifically, S anions were found to adjust the electronic structure and morphology of the material, while according to their DFT calculations Ag^+ cations



Table 1 Performance summary of recently reported copper sulfide eCO₂RR catalysts where partial current densities were not provided in the literature they were calculated by multiplying the total current density with the Faradaic efficiency of the main product

Preparation method	Catalysts	Electrolyte	<i>E</i> (V vs. RHE)	Partial current density (mA cm ⁻²)	Product (FE)	Ref.	
Solvothermal	Cu ₂ O/CuS Nanocomposites	0.1 M KHCO ₃	−0.9	15.3	HCOO [−] (67.6%)	74	
	N-Enriched CuS	0.1 M KHCO ₃	−0.8	NA	HCOO [−] (~88%)	81	
	Copper sulfide nanocrystals	0.1 M KHCO ₃	−1	2.1	HCOO [−] (52.2%)	82	
	CuS@SnO ₂ nanosheets	0.5 M KHCO ₃	−1.0	~150	HCOO [−] (83.4%)	73	
	CuS@MoS ₂ heterostructures	CO ₂ saturated electrolyte	−0.69	NA	NA	68	
	CuS/N, S-rGO	0.5 M KHCO ₃	−0.63	~5	HCOO [−] (82%)	71	
	CuS-thiourea	0.1 M KHCO ₃	−0.51	~0.8	CO (72.67%)	80	
	Cu ₂ S nanocrystal-decorated Cu nanosheets	0.1 M KHCO ₃	−1.2	~20.7	C ₂ H ₅ OH (46%)	78	
	Sm _x -CuS _y	1 M KOH	−0.52	276	HCOO [−] (92.1%)	77	
	Sulfur tailored Cu ₂ O	0.1 M KHCO ₃	−1	16.3	HCOO [−] (67.2%)	70	
Electrosynthesis	Active CuS _x	0.1 M KHCO ₃	−0.9	9	HCOO [−] (75%)	90	
	Sulfide-derived Cu	0.1 M KHCO ₃	−0.8	~2.5	HCOOH (~60%)	91	
	Cu ₂ S _{1−x} with abundant Cu ^{δ+}	0.5 M KHCO ₃	−0.3	~2.5	C ₂ H ₅ OH (73.3%)	93	
	Dual-doped Ag, S-Cu ₂ O/Cu	BMIMBF ₄ /H ₂ O	−1.18	82.7	CH ₃ OH (67.4%)	94	
Cation exchange	CuS _x	0.1 M KHCO ₃	−0.6	2.2	HCOO [−] (71.8%)	62	
	3D Cu ₂ S	0.1 M NaHCO ₃	−0.9	19.1	HCOO [−] (87.3%)	96	
	Ag/Cu sulfide	0.05 M K ₂ CO ₃	−1.2	5.1	C ₂ H ₄ (34%)	97	
Wet chemistry	Cu and CuS nano-particles@graphene nanoflakes	0.1 M KHCO ₃	−0.6	2.3	HCOO [−] (42.5%)	79	
	SnO ₂ confined on CuS nanosheets	0.1 M KHCO ₃	−0.8	~17	syngas (> 85%)	72	
	Sulfide-derived CuCd	0.1 M KHCO ₃	−0.8	0.6	C ₂ H ₅ OH (~32%)	76	
	Sulfur vacancy-rich CuS	0.1 M KHCO ₃	−1.05	~3.3	<i>n</i> -Propanol (15.4%)	75	
	Hierarchical CuS hollow polyhedrons	0.5 M K ₂ SO ₄	−0.6	~16	HCOO [−] (~50%)	69	
Other methods	Sulfur-modified copper catalysts	0.1 M KHCO ₃	−0.8	~21	HCOO [−] (80%)	67	
	Thermal oxidation	CuS decorated CuO heterostructure	0.1 M KHCO ₃	−0.7	20	HCOO [−] (84%)	95
	<i>In situ</i> crystallization	Cu _{1.81} S particles on carbon nanotubes	0.5 M KHCO ₃	−0.67	3.1	HCOO [−] (82%)	60
	Chemical bath deposition	CuS nanosheet arrays	0.5 M KHCO ₃	−0.7	~50	HCOO [−] (70.2%)	61
	Impregnation and annealing	Cu _{2−x} S derived nanoparticles@C and/or 0.5 M KCl	0.5 M KHCO ₃	−0.78	0.18	HCOO [−] (12%)	65

suppressed HER. Zhang *et al.*⁶⁹ studied the eCO₂RR catalytic properties of CuS hollow polyhedra that experience structural transformation, as they evolve into a sulfur-doped metallic Cu catalyst. The *in situ* evolved catalyst exhibited a high durability and formate selectivity, maintaining stable operation for 36 hours at a formate partial current density of approximately 16 mA cm^{−2} (at −0.6 V vs. RHE). This stability and selectivity were underscored by DFT calculations, which highlighted the significant role of sulfur-doped Cu (111) facets in its performance.

3.2 Synergistic effects of heterostructures and hybrid materials

Advancements in material science have demonstrated that heterostructures, comprising multilayer materials, and hybrid materials allow to optimize the catalytic CO₂RR performance, selectivity, and stability through synergistic effects.^{68,71–74,76–78,95} For example, CuS-based heterostructures with other semiconductors or metal catalysts can have optimized charge transfer, energy band alignment, and beneficial interfacial interactions to enhance their eCO₂RR performance. The same applies to hybrid materials that combine CuS with conductive polymers, metal-organic frameworks, or graphene-based materials.

Wang *et al.*⁷³ discovered that Cu₂SnS₃ nanosheets undergo a self-adaptive phase separation during the electrochemical process, resulting in the formation of stable SnO₂@CuS and SnS₂@Cu₂O heterojunctions. Because Sn⁴⁺ cations can be easily drawn by the negative potential, they migrate from the core to the surface of the nanosheet, replacing coordinated S atoms with O to form SnO₂ nanoparticles. Meanwhile, Cu⁺ ions, carrying S atoms, concentrate in the center, forming CuS (Fig. 7). They identified that activated Sn is responsible for CO₂ reduction in CuS@SnO₂ ternary sulfides, which facilitates a self-adaptive transformation. By harnessing the synergistic effect of a stable heterojunction structure and electron self-flow, the optimized catalysts efficiently convert CO₂ to formate with 84.1% FE for 10 hours. This is supported by DFT calculations demonstrating that at the SnO₂ and CuS/Cu₂O interface, delocalized Sn⁴⁺ electrons transfer to Cu⁺, thereby enhancing the HCOO* affinity, promoting H₂O dissociation, and stabilizing Sn⁴⁺ sites, which collectively improve the CO₂RR activity and selectivity. This work revealed the active center of the ternary sulfide, offering insights for designing better electrocatalysts. Wang *et al.*⁷⁴ presented an innovative Cu₂O/CuS composite catalyst that exhibits a 67.6% FE and a notable partial current density of 15.3 mA cm^{−2} at −0.9 V *versus* RHE for formate.



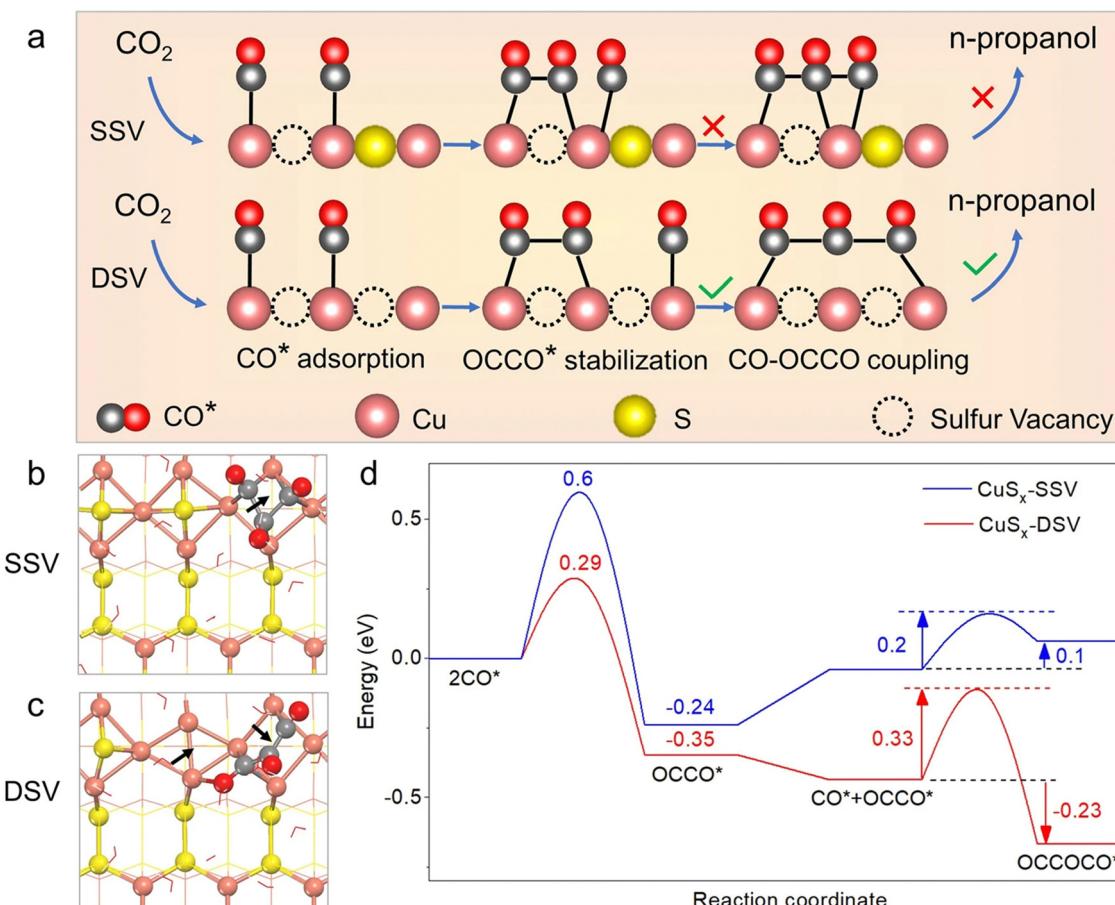


Fig. 6 (a) Schematic illustration of the conversion of CO₂ to n-propanol on CuS_x with double sulfur vacancies (DSV) and related calculations. The process involves n-propanol formation on adjacent CuS_x-DSV through CO–CO dimerization and CO–OCCO coupling. Structural diagrams displaying the optimized OCCOCO* intermediate configurations on the (100) surface of CuS_x with (b) a single sulfur vacancy (SSV) and (c) CuS_x-DSV. Arrows indicate sulfur vacancy positions. (d) Energy diagrams for CuS_x-SSV (blue curve) and CuS_x-DSV (red curve) at 0 V vs. RHE. Copyright 2021 Nature Publishing Group.⁷⁵

This catalyst maintains an average FE of 62.9% for at least 30 hours at the same potential, outperforming other Cu, CuS, and Cu₂O catalysts in terms of selectivity and stability. The incorporation of CuS is believed to stabilize Cu₂O during CO₂ reduction. DFT calculations imply that the CuS (110) surface favors the formation of formate over CO. In another study, Wang *et al.*⁷² designed a novel hierarchical structure combining SnO₂ nanoparticles with CuS nanosheets, improving the electroreduction of CO₂ into a versatile syngas mixture (CO/H₂ ratio varying in the range of 0.11–3.86). The material achieved a remarkable FE >85%, a significant turnover frequency of 96.1 h⁻¹, and a durability of more than 24 hours. Experimental characterization and theoretical calculations demonstrated that various factors like size, morphology and the density of the SnO₂–CuS interface contribute to the enhanced catalytic performance. In a study by Mosali *et al.*,⁷⁶ a near 30% FE for ethanol production for 6 hours was reported (at -0.8 V *versus* RHE), using the sulfide-derived-CuCd₂ catalyst in 0.1 M KHCO₃. This notable ethanol selectivity was attributed to the abundant distribution of CuS/Cu₂S–CdS boundaries and Cu₂S sites within the sulfide-derived-CuCd₂ catalyst identified through electrochemical, spectroscopic, and

microscopic analyses. The electrochemically stable CdS phase plays a crucial role in creating the favorable catalyst structure, which stimulated C–C coupling and further reduction to ethanol. Li *et al.*⁷⁸ investigated an electroreductive CO₂-to-ethanol conversion process, catalyzed by Cu nanosheets that are uniformly decorated with small Cu₂S nanocrystals. The system, housed in an H-cell containing 0.1 M KHCO₃, achieved a 20-hour stable high ethanol current density of approximately 20.7 mA cm⁻² with a large FE of 46% (at 1.2 V *versus* RHE). Nanocrystal-decorated Cu nanosheets offer several advantages over heteroatom doping, Cu sulfide/oxide/nitride, and the parent catalyst, including: (i) a relatively high positive local charge of Cu^{δ+}; (ii) effective interfaces between Cu^{δ+} and Cu⁰; and (iii) a non-flat, stepped surface. These features improve CO₂ adsorption, enhance eCO₂RR kinetics, facilitate the dissociation of the *COOH intermediate into *CO, strengthen the affinity of *CO on the catalyst, and bring the energy barrier for C–C coupling below the desorption energy of *CO. Liu *et al.*⁷⁷ found that incorporating Sm₂O₃ into an S-doped Cu matrix creates a mixed phase of heterogeneous Sm and homogeneous S co-doped Cu, which stabilizes Cu^{δ+} species and encourages the HCOOH formation pathway. With the

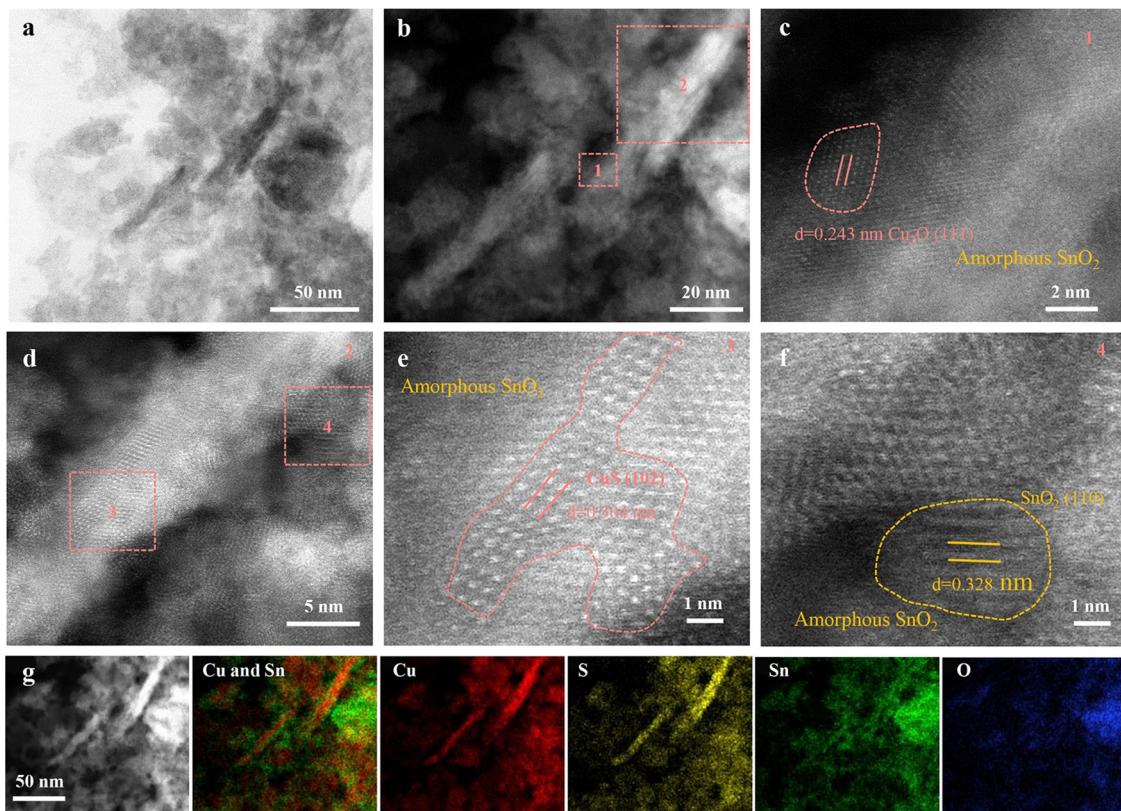


Fig. 7 Characterization of Cu_2SnS_3 after CO_2RR (R- Cu_2SnS_3). (a) TEM image. (b)–(f) HAADF-STEM images. (g) HAADF-STEM image and EDX element mapping. Copyright 2021 Wiley-VCH.⁷³

$\text{Sm}_{0.06}\text{-CuS}_{0.03}$ catalyst, they achieved a 92.1% FE for HCOOH production for 24 hours and an impressive current density of 300 mA cm^{-2} at a low reduction potential of -0.52 V vs. RHE. This good electrocatalytic performance could, based on material characterizations and *in situ* experiments, be attributed to synergistic interaction between heterogeneously doped Sm species and homogeneously doped S elements. To explore the synergy between Sm and S, homogeneous S doping in the Cu active phase has been studied. This modification alters the inherent electronic structure of Cu, generating a surplus of $\text{Cu}^{\delta+}$ ($\delta = +0.61$). The presence of heterogeneous Sm species not only stabilizes $\text{Cu}^{\delta+}$ sites but also raises the concentration of H ions, which enhances the adsorption of oxygen-containing intermediates ($^*\text{OCHO}$) during HCOOH formation.

3.3 Size and morphology engineering

This section comments on recent approaches to customize CuS ^{60,62,67,70,80,82,96} catalyst properties by particle size and morphology engineering. Through the optimization of these aspects, researchers continue to unveil new opportunities of copper sulfides in electrochemical CO_2 reduction.

Shinagawa *et al.*⁶⁷ identified a relationship between the particle size and the formate selectivity in sulfur-modified copper catalysts. The particle size varies between fresh and used samples. After the electrochemical test, the initial average particle size of about 15 ± 7 nm for a particular CuS samples reduced to approximately 3 nm, while that of another sample

decreased from roughly 42 nm to around 5 nm. They found that CuS -4 with larger particle sizes and higher sulfur contents resulted in increased FE for formate. Submicron-sized CuS catalysts, prepared solvothermally, outperformed their nano-sized counterparts. In regard to stability, we note that the Cu-S catalyst sustained its exclusive formate production over 12 hours at -0.8 V vs. RHE. He *et al.*⁹⁶ uncovered a relationship between electrochemical parameters like the applied potential and charge passed and the extent of electrochemically-driven cation exchange reactions through structural and compositional analyses. This allowed them to systematically adjust binary cation ratios in nanocrystalline films, while preserving their initial 3D structural characteristics and engineer highly reactive catalytic sites by generating ultra-small structures and controlling surface oxidation states. Using this approach, they synthesized a Cu_2S electrocatalyst with high selectivity and activity for eCO_2RR , achieving a CO_2 partial current density over 19 mA cm^{-2} for formate production at about -0.9 V vs. RHE with a FE of 87.3% (stable for 9 hours at -0.82 V vs. NHE). This performance exceeds that of all Cu-based catalysts previously reported, providing insights into targeting specific reactive sites within nanocrystalline metal sulfides for eCO_2RR and suggesting a method for designing advanced electrocatalysts for various energy conversion processes. In their research, Lim *et al.*⁶² developed highly selective and stable copper sulfide (CuS_x) catalysts capable of converting industrial CO_2 into formate. The copper foil spontaneously reacted with sulfur

species, increasing the average size and surface density of the CuS_x nanoparticles as the H_2S concentration rose, resulting in an increase of the formate FE from 22.7% to 71.8%. CuS_x catalysts demonstrated efficacy in both industrial and pure CO_2 saturated electrolytes and maintained stability for 12 hours of continuous operation. The high FE for formate achieved by the CuS_x catalysts was due to the significant sulfur content, enhanced size and augmented surface density of CuS_x nanoparticles. Gao *et al.*⁸⁰ investigated the synthesis of CuS catalysts using various sulfur precursors for electrochemical CO_2 reduction, and found that these precursors influence the catalyst performance due to different morphological features and CO selectivity. The CuS thiourea catalyst, with its distinctive morphology resembling nanoflowers, outperformed others by achieving the highest CO FE of 72.67% and favorable CO_2 reduction kinetics.

Generally, the strategic engineering of catalysts' size and morphology selectively enhances CO_2 reduction, while minimizing HER. This approach entails increasing exposure of boundary density and active sites, controlling exposed catalyst facets, and adjusting particle sizes. Furthermore, a refined morphology facilitates better mass transfer and encourages favorable kinetics for CO_2 reduction reactions.

4. Recent strategies to enhance the performance of copper selenide in eCO_2RR

Similar as for CuS, recent strategies to improve the performance of copper selenide in the electrochemical CO_2 reduction include structural and compositional tuning,^{64,83–85,87,88,103,104} size and morphology engineering,^{83,89,92} and heterostructure design.⁸⁶ Table 2 provides a compact summary of the latest results obtained on CuSe-based eCO_2RR catalysts, detailing their composition, preparation method, performance and product selectivity according to the electrolyte used. A comparison

of the CO_2 electroreduction efficiency of copper chalcogenides prepared by alternative methods to those employed for the current leading catalysts is depicted in Fig. 9.

The most common strategy for designing better catalysts is to optimize the catalyst's structure and composition. Wang *et al.*⁸⁵ designed bimetallic CuInSe_2 with Se vacancies to mimic the eCO_2RR catalytic properties of Au, resulting in an efficient conversion to CO accompanied by a suppressed HER activity. The orbital interaction of both metals with Se vacancies played a critical role in the eCO_2RR pathway. Se-defective CuInSe_2 (V-CuInSe_2) nanosheets exhibited an effective performance, illustrating the potential of cost-effective, noble-metal-like catalysts with unique electronic, structural, and catalytic properties. Duan *et al.*⁸⁸ suggested using natural superlattices with alternating active/conductive layers to stabilize the metal oxidation states, improving the eCO_2RR performance. They utilized BiCuSeO that features alternating conductive $[\text{Cu}_2\text{Se}_2]^{2-}$ and insulating $[\text{Bi}_2\text{O}_2]^{2+}$ sublayers as a representative example. It was demonstrated that the bismuth oxide layers played a significant role in activating CO_2 , with created electrons being transferred quickly through the $[\text{Cu}_2\text{Se}_2]^{2-}$ sublayers. DFT computations suggested that the Bi–O coordination in $[\text{Bi}_2\text{O}_2]^{2+}$ effectively activates and stabilizes OCHO^* intermediates, which facilitates the selective CO_2 electroreduction to formate. As a result, the naturally occurring BiCuSeO superlattices generate formate with a FE over 90% across a broad potential range between -0.4 and -1.1 V in a neutral electrolyte. This work demonstrates the use of natural superlattices for enhanced eCO_2RR selectivity and highlights the importance of specific coordination structures (see design strategies in Fig. 8). Ding *et al.*⁸⁷ found that exposed (220) crystal planes facilitate the adsorption of a key bridge-bonded $^*\text{CO}_\text{B}$ intermediate. Potassium doping enhances the adsorption of linear $^*\text{CO}_\text{L}$ and bridge $^*\text{CO}_\text{B}$ intermediates on the catalyst surface, promoting C–C coupling to produce ethanol by protecting Cu(i) sites during the electrocatalysis. The optimal K11.2%– Cu_2Se nanosheet array achieves a 70.3% FE and a partial current density of 35.8 mA cm^{-2} for ethanol at -0.8 V in 0.1 M KHCO_3 , while remaining stable for

Table 2 Performance summary of recently reported copper selenide eCO_2RR catalysts where partial current densities were not provided in the literature they were calculated by multiplying the total current density with the Faradaic efficiency of the main product

Preparation method	Catalysts	Electrolyte	<i>E</i> (V vs. RHE)	Partial current density (mA cm ⁻²)	Product and FE	Ref.
Solothermal	BiCuSeO superlattices	0.5 M KHCO_3	−0.9	18.7	HCOO^- (~93.4%)	88
	$\text{Cu}_{1.63}\text{Se}$	[BMIM]PF ₆ (30 wt%)/ $\text{CH}_3\text{CN}/\text{H}_2\text{O}$ (5 wt%)	−2.1 V vs. Ag/Ag ⁺	32.2	CH_3OH (77.6%)	83
	Se-defective CuInSe_2	0.5 M KHCO_3	−0.7	112	CO (91%)	85
	Nanostructured Cu_2Se	0.3 M NaHCO_3	−1.3	NA	HCOO^- (94.2%)	84
	2D $\text{CuSe}/\text{g-C}_3\text{N}_4$	0.1 M KHCO_3	−1.2	~8.4	CO (85.28%)	86
	K11.2%– Cu_2Se nanosheets	0.1 M KHCO_3	−0.8	35.8	$\text{C}_2\text{H}_5\text{OH}$ (70.3%)	87
	$\text{Cu}_{1.22}\text{V}_{0.19}\text{Se}$ nanotubes	0.1 M KHCO_3	−0.8	207.9	$\text{C}_2\text{H}_5\text{OH}$ (68.3%)	105
	Cu_3Se_2 nanosheet on Cu foam	0.1 M KHCO_3	−1.2	11.9	CO (>40%)	89
Wet chemistry and annealing	2D Cu_{2-x}Se with numerous Se vacancies	0.5 M KHCO_3	−0.8	7.44	$\text{C}_2\text{H}_5\text{OH}$ (68.1%)	106
	$\text{Cu}_{1.82}\text{Se}$ nanowires	0.1 M KHCO_3	−1.1	10.2	C_2H_4 (55%)	64
Electro-synthesis	N-CuSe	0.5 M KHCO_3	−1.1	25.8	HCOO^- (55.2%)	104
	3D CuSe nanocubes	1.0 M KHCO_3 or 0.1 M [BMIM]PF ₆	−1.3	10	NA	92



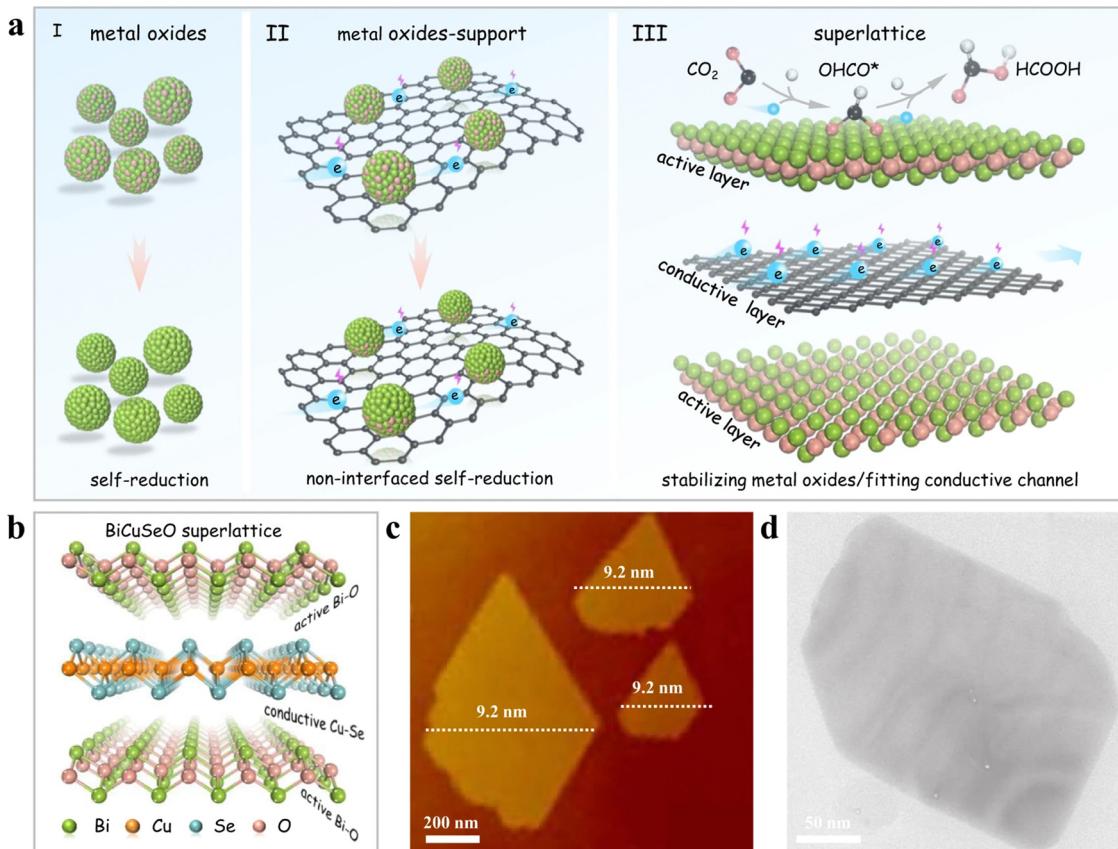


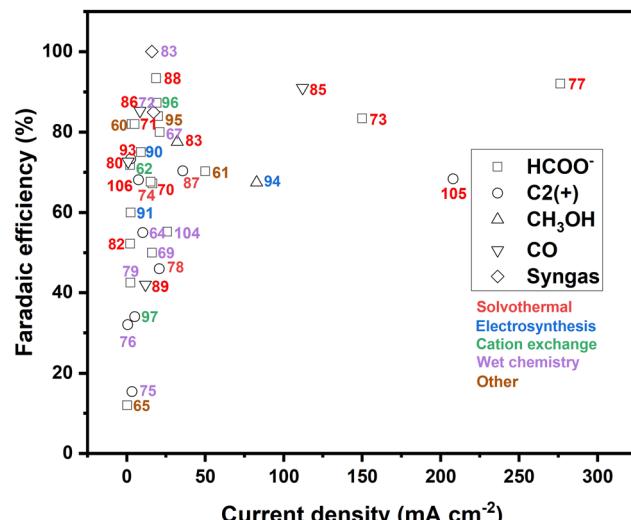
Fig. 8 Development of BiCuSeO superlattice nanosheets: design strategy and structural analysis. (a) Visual representation of single-phase materials using alternating active and conductive layer arrangements, including (I) pure metal oxides, (II) metal oxides interacting with a support, and (III) a superlattice made up of vertically stacked active (metal oxides) and conductive CuSe layers in a one-to-one ratio. (b) Structural representation of the BiCuSeO superlattice, (c) AFM image, and (d) TEM image. Copyright 2022 Nature Publishing Group.⁸⁸

130 hours. With a FE of ethanol exceeding 50% in a wide potential range from -0.6 to -1.2 V, this approach demonstrates the good potential of these superlattices for real-world implementations. This study presents a novel method to enhance CO_2 RR to ethanol through alkali metal doping and offers a rational strategy for designing catalysts with high product selectivity for multiple proton-coupled electron transfer reactions. Sun *et al.*¹⁰⁵ developed V-doped Cu_2Se nanotubes for flow-cell CO_2 RR, efficiently producing ethanol. The introduction of V^{4+} ions into Cu_2Se provides diverse active sites, prevents Cu^+ reduction, and facilitates different CO intermediates' adsorption, leading to ethanol formation, as confirmed by *in situ* DRIFTS and DFT calculations. The optimized $\text{Cu}_{1.22}\text{V}_{0.19}\text{Se}$ nanotubes achieve a 68.3% $\text{FE}_{\text{C}_2\text{H}_5\text{OH}}$ and a -207.9 mA cm^{-2} partial current density for ethanol at -0.8 V in 1 M KOH. Hao *et al.*⁸⁹ presented self-supported Cu_3Se_2 featuring a hierarchical structure composed of nanosheet-assembled fibers whose properties are modulated by anions. This structure delivers a flexible and broad range of H_2 -to-CO ratios (from 0.8 to 6.0) within an extensive potential window and boasts an impressive turnover frequency (TOF) of 1303 h $^{-1}$. Wang *et al.*¹⁰⁶ prepared 2D Cu_{2-x}Se with numerous Se vacancies, enhancing CO_2 to ethanol conversion activity and selectivity during e CO_2 RR. The optimal Cu-Cu spacing in this ultra-thin

material, obtained by the presence of Se vacancies, promotes C-C coupling, resulting in efficient ethanol production. Specifically, $\text{V}_{\text{Se}}\text{-Cu}_{2-x}\text{Se}$, with a 2.51 Å Cu-Cu distance, achieves a 68.1% faradaic efficiency and 7.44 mA cm^{-2} partial current density for ethanol in 0.5 M KHCO_3 , within a -0.4 to -1.6 V potential range. *In situ* Fourier transform infrared spectroscopy and DFT calculations confirm that this spacing optimizes the Gibbs free energy for both ethanol generation and formate formation steps.

Yang *et al.*⁸³ emphasized that the copper selenide catalysts' performance was strongly influenced by their size and morphology, which could be fine-tuned using the volume ratio of diethylenetriamine and water (VDETA/VH₂O). They found that the catalysts synthesized at a VDETA/VH₂O ratio of 1/3 had the smallest size, resulting in the highest current density and FE. The $\text{Cu}_{1.63}\text{Se}(1/3)$ nanocatalysts achieved an exceptional current density of 41.5 mA cm^{-2} and an FE of 77.6% at a potential of -2.1 V *versus* Ag/Ag^+ .

Since the first application of copper selenide in the electrocatalytic carbon dioxide reduction in 2019,⁸³ the field has seen substantial progress. This advancement is highlighted by the diversification of the reaction products that now include methanol, formic acid, carbon monoxide (CO), and notably, ethanol.^{87,105,106} Additionally, there has been a remarkable



2 T. K. Todorova, M. W. Schreiber and M. Fontecave, *ACS Catal.*, 2019, **10**, 1754–1768.

3 G. M. Tomboc, S. Choi, T. Kwon, Y. J. Hwang and K. Lee, *Adv. Mater.*, 2020, **32**, e1908398.

4 L. Zhang, I. Merino-Garcia, J. Albo and C. M. Sánchez-Sánchez, *Curr. Opin. Electrochem.*, 2020, **23**, 65–73.

5 J. Zhao, S. Xue, J. Barber, Y. Zhou, J. Meng and X. Ke, *J. Mater. Chem. A*, 2020, **8**, 4700–4734.

6 S. Bhardwaj, A. Biswas, M. Das and R. S. Dey, *Sustain. Energy Fuels.*, 2021, **5**, 2393–2414.

7 Y. Jia, F. Li, K. Fan and L. Sun, *Adv. Powder Mater.*, 2022, **1**, 100012.

8 D. Karapinar, C. E. Creissen, J. G. Rivera de la Cruz, M. W. Schreiber and M. Fontecave, *ACS Energy Lett.*, 2021, **6**, 694–706.

9 K. Van Daele, B. De Mot, M. Pupo, N. Daems, D. Pant, R. Kortlever and T. Breugelmans, *ACS Energy Lett.*, 2021, **6**, 4317–4327.

10 O. Gutiérrez-Sánchez, B. de Mot, N. Daems, M. Bulut, J. Vaes, D. Pant and T. Breugelmans, *Energy Fuels*, 2022, **36**, 13115–13123.

11 J. Mertens, C. Breyer, K. Arning, A. Bardow, R. Belmans, A. Dibenedetto, S. Erkman, J. Gripekoven, G. Léonard, S. Nizou, D. Pant, A. S. Reis-Machado, P. Styring, J. Vente, M. Webber and C. J. Sapart, *Joule*, 2023, **7**, 442–449.

12 C. Liu, J. Gong, Z. Gao, L. Xiao, G. Wang, J. Lu and L. Zhuang, *Sci. China: Chem.*, 2021, **64**, 1660–1678.

13 W. Ma, X. He, W. Wang, S. Xie, Q. Zhang and Y. Wang, *Chem. Soc. Rev.*, 2021, **50**, 12897–12914.

14 W. Quan, Y. Lin, Y. Luo and Y. Huang, *Adv. Sci.*, 2021, **8**, 2101597.

15 Z. Sun, Y. Hu, D. Zhou, M. Sun, S. Wang and W. Chen, *ACS Energy Lett.*, 2021, **6**, 3992–4022.

16 B. Talukdar, S. Mendiratta, M. H. Huang and C. H. Kuo, *Chem. – Asian J.*, 2021, **16**, 2168–2184.

17 C. Xiao and J. Zhang, *ACS Nano*, 2021, **15**, 7975–8000.

18 Y. Xin, K. Yu, L. Zhang, Y. Yang, H. Yuan, H. Li, L. Wang and J. Zeng, *Adv. Mater.*, 2021, **33**, e2008145.

19 B. Deng, X. Zhao, Y. Li, M. Huang, S. Zhang and F. Dong, *Sci. China: Chem.*, 2022, **66**, 78–95.

20 H. Li, P. Wei, D. Gao and G. Wang, *Curr. Opin. Green Sustain. Chem.*, 2022, **34**, 100589.

21 S. Mohan, B. Honnappa, A. Augustin, M. Shanmugam, C. Chuaicham, K. Sasaki, B. Ramasamy and K. Sekar, *Catalysts*, 2022, **12**, 445.

22 S. Mu, H. Lu, Q. Wu, L. Li, R. Zhao, C. Long and C. Cui, *Nat. Commun.*, 2022, **13**, 3694.

23 L. Zaza, K. Rossi and R. Buonsanti, *ACS Energy Lett.*, 2022, **7**, 1284–1291.

24 Z. Zhang, L. Bian, H. Tian, Y. Liu, Y. Bando, Y. Yamauchi and Z. L. Wang, *Small*, 2022, **18**, 2107450.

25 C. Zhu, S. Zhao, G. Shi and L. Zhang, *ChemSusChem*, 2022, **15**, e202200068.

26 Y. Lei, Z. Wang, A. Bao, X. Tang, X. Huang, H. Yi, S. Zhao, T. Sun, J. Wang and F. Gao, *J. Chem. Eng.*, 2023, **453**, 139663.

27 Y. Yang, S. Louisia, S. Yu, J. Jin, I. Roh, C. Chen, M. V. Fonseca Guzman, J. Feijoo, P. C. Chen, H. Wang, C. J. Pollock, X. Huang, Y. T. Shao, C. Wang, D. A. Muller, H. D. Abruna and P. Yang, *Nature*, 2023, **614**, 262–269.

28 C. Choi, S. Kwon, T. Cheng, M. Xu, P. Tieu, C. Lee, J. Cai, H. M. Lee, X. Pan, X. Duan, W. A. Goddard and Y. Huang, *Nat. Catal.*, 2020, **3**, 804–812.

29 J. Bu, Z. Liu, W. Ma, L. Zhang, T. Wang, H. Zhang, Q. Zhang, X. Feng and J. Zhang, *Nat. Catal.*, 2021, **4**, 557–564.

30 C. Durante, *Nat. Catal.*, 2021, **4**, 537–538.

31 C. Kim, J. C. Bui, X. Luo, J. K. Cooper, A. Kusoglu, A. Z. Weber and A. T. Bell, *Nat. Energy*, 2021, **6**, 1026–1034.

32 F. Li, X. V. Medvedeva, J. J. Medvedev, E. Khairullina, H. Engelhardt, S. Chandrasekar, Y. Guo, J. Jin, A. Lee, H. Thérien-Aubin, A. Ahmed, Y. Pang and A. Klinkova, *Nat. Catal.*, 2021, **4**, 479–487.

33 M. C. O. Monteiro, F. Dattila, B. Hagedoorn, R. García-Muelas, N. López and M. T. M. Koper, *Nat. Catal.*, 2021, **4**, 654–662.

34 R. Shi, Z. Wang, Y. Zhao, G. I. N. Waterhouse, Z. Li, B. Zhang, Z. Sun, C. Xia, H. Wang and T. Zhang, *Nat. Catal.*, 2021, **4**, 565–574.

35 H. Shin, K. U. Hansen and F. Jiao, *Nat. Sustain.*, 2021, **4**, 911–919.

36 Z. Xing, L. Hu, D. S. Ripatti, X. Hu and X. Feng, *Nat. Commun.*, 2021, **12**, 136.

37 T. Zheng, C. Liu, C. Guo, M. Zhang, X. Li, Q. Jiang, W. Xue, H. Li, A. Li, C. W. Pao, J. Xiao, C. Xia and J. Zeng, *Nat. Nanotechnol.*, 2021, **16**, 1386–1393.

38 Z. Han, D. Han, Z. Chen, J. Gao, G. Jiang, X. Wang, S. Lyu, Y. Guo, C. Geng, L. Yin, Z. Weng and Q. H. Yang, *Nat. Commun.*, 2022, **13**, 3158.

39 J. Timoshenko, A. Bergmann, C. Rettenmaier, A. Herzog, R. M. Arán-Ais, H. S. Jeon, F. T. Haase, U. Hejral, P. Grosse, S. Kühl, E. M. Davis, J. Tian, O. Magnussen and B. Roldan Cuenya, *Nat. Catal.*, 2022, **5**, 259–267.

40 X. Wang, P. Ou, A. Ozden, S.-F. Hung, J. Tam, C. M. Gabardo, J. Y. Howe, J. Sisler, K. Bertens, F. P. García de Arquer, R. K. Miao, C. P. O'Brien, Z. Wang, J. Abed, A. S. Rasouli, M. Sun, A. H. Ip, D. Sinton and E. H. Sargent, *Nat. Energy*, 2022, **7**, 170–176.

41 Y. Xie, P. Ou, X. Wang, Z. Xu, Y. C. Li, Z. Wang, J. E. Huang, J. Wicks, C. McCallum, N. Wang, Y. Wang, T. Chen, B. T. W. Lo, D. Sinton, J. C. Yu, Y. Wang and E. H. Sargent, *Nat. Catal.*, 2022, **5**, 564–570.

42 A. Xu, S.-F. Hung, A. Cao, Z. Wang, N. Karmodak, J. E. Huang, Y. Yan, A. Sedighian Rasouli, A. Ozden, F.-Y. Wu, Z.-Y. Lin, H.-J. Tsai, T.-J. Lee, F. Li, M. Luo, Y. Wang, X. Wang, J. Abed, Z. Wang, D.-H. Nam, Y. C. Li, A. H. Ip, D. Sinton, C. Dong and E. H. Sargent, *Nat. Catal.*, 2022, **5**, 1081–1088.

43 T. Zhang, J. C. Bui, Z. Li, A. T. Bell, A. Z. Weber and J. Wu, *Nat. Catal.*, 2022, **5**, 202–211.

44 B. Seger, M. Robert and F. Jiao, *Nat. Sustain.*, 2023, **6**, 236–238.



45 D. R. Brown, T. Day, K. A. Borup, S. Christensen, B. B. Iversen and G. J. Snyder, *APL Mater.*, 2013, **1**, 052107.

46 C. Han, Q. Sun, Z. Li and S. X. Dou, *Adv. Energy Mater.*, 2016, **6**, 1600498.

47 Y. Shi, C. Sturm and H. Kleinke, *J. Solid State Chem.*, 2019, **270**, 273–279.

48 T. Deng, T. Xing, M. K. Brod, Y. Sheng, P. Qiu, I. Veremchuk, Q. Song, T.-R. Wei, J. Yang, G. J. Snyder, Y. Grin, L. Chen and X. Shi, *Energy Environ. Sci.*, 2020, **13**, 3041–3053.

49 J. Jiang, H. Zhu, Y. Niu, Q. Zhu, S. Song, T. Zhou, C. Wang and Z. Ren, *J. Mater. Chem. A*, 2020, **8**, 4790–4799.

50 M. M. Kubanova, K. A. Kuterbekov, M. K. Balapanov, R. K. Ishembetov, A. M. Kabyshev and K. Z. Bekmyrza, *Nanomater.*, 2021, **11**, 2238.

51 J. R.-F. Ilka Kriegel, A. Wisnet, H. Zhang, C. Waurisch, A. Eychmüller, A. Dubavik, A. O. Govorov and J. Feldmann, *ACS Nano*, 2013, **7**, 4367–4377.

52 B. Hamawandi, S. Ballikaya, M. Rasander, J. Halim, L. Vinciguerra, J. Rosen, M. Johnsson and M. Toprak, *Nanomater.*, 2020, **10**, 854.

53 T. Willhammar, K. Sentosun, S. Mourdikoudis, B. Goris, M. Kurttepeli, M. Bercx, D. Lamoen, B. Partoens, I. Pastoriza-Santos, J. Perez-Juste, L. M. Liz-Marzan, S. Bals and G. Van Tendeloo, *Nat. Commun.*, 2017, **8**, 14925.

54 X. Chen, J. Yang, T. Wu, L. Li, W. Luo, W. Jiang and L. Wang, *Nanoscale*, 2018, **10**, 15130–15163.

55 W. Li, A. Shavel, R. Guzman, J. Rubio-Garcia, C. Flox, J. Fan, D. Cadavid, M. Ibanez, J. Arbiol, J. R. Morante and A. Cabot, *Chem. Commun.*, 2011, **47**, 10332–10334.

56 Y. Zhao and C. Burda, *Energy Environ. Sci.*, 2012, **5**, 5564–5576.

57 C. Coughlan, M. Ibanez, O. Dobrozhana, A. Singh, A. Cabot and K. M. Ryan, *Chem. Rev.*, 2017, **117**, 5865–6109.

58 B. Yun, H. Zhu, J. Yuan, Q. Sun and Z. Li, *J. Mater. Chem. B*, 2020, **8**, 4778–4812.

59 Y. Chen, K. Chen, J. Fu, A. Yamaguchi, H. Li, H. Pan, J. Hu, M. Miyauchi and M. Liu, *Nano Mater. Sci.*, 2020, **2**, 235–247.

60 B. Zhang, M. Wang, J. Ding, Y. Li, G. Cao, M. T. Bernards, Y. He and Y. Shi, *J. CO₂ Util.*, 2020, **39**, 101169.

61 T. Dou, Y. Qin, F. Zhang and X. Lei, *ACS Appl. Energy Mater.*, 2021, **4**, 4376–4384.

62 J. W. Lim, W. J. Dong, J. Y. Park, D. M. Hong and J. L. Lee, *ACS Appl. Mater. Interfaces*, 2020, **12**, 22891–22900.

63 K. Sun, S. Song, A. Kumar, Z. Shang, X. Duan, S. Wang, S. Tian, J. Tang, D. Zhou and X. Sun, *ACS Appl. Nano Mater.*, 2021, **4**, 2357–2364.

64 Y. Mi, X. Peng, X. Liu and J. Luo, *ACS Appl. Energy Mater.*, 2018, **1**, 5119–5123.

65 C. H. M. van Oversteeg, M. Tapia Rosales, K. H. Helfferich, M. Ghiasi, J. D. Meeldijk, N. J. Firet, P. Ngene, C. de Mello Donegá and P. E. de Jongh, *Catal. Today*, 2021, **377**, 157–165.

66 J. Xu, W. Zhang, Z. Yang, S. Ding, C. Zeng, L. Chen, Q. Wang and S. Yang, *Adv. Funct. Mater.*, 2009, **19**, 1759–1766.

67 T. Shinagawa, G. O. Larrazábal, A. J. Martín, F. Krumeich and J. Pérez-Ramírez, *ACS Catal.*, 2018, **8**, 837–844.

68 F. Liu, R. Xu, L. Hong, Z. Yang, Y. Li, C. Wang, H. Jia and C. Yang, *ChemistrySelect*, 2020, **5**, 360–368.

69 X. Zhang, R. Sa, F. Zhou, Y. Rui, R. Liu, Z. Wen and R. Wang, *CCS Chem.*, 2021, **3**, 199–207.

70 H. Yuan, Z. Liu, S. Sang and X. Wang, *Appl. Surf. Sci.*, 2023, **613**, 156130.

71 Z. Wu, J. Yu, K. Wu, J. Song, H. Gao, H. Shen, X. Xia, W. Lei and Q. Hao, *Appl. Surf. Sci.*, 2022, **575**, 151796.

72 X. Wang, J. Lv, J. Zhang, X. L. Wang, C. Xue, G. Bian, D. Li, Y. Wang and T. Wu, *Nanoscale*, 2020, **12**, 772–784.

73 W. Wang, Z. Wang, R. Yang, J. Duan, Y. Liu, A. Nie, H. Li, B. Y. Xia and T. Zhai, *Angew. Chem., Int. Ed.*, 2021, **60**, 22940–22947.

74 S. Wang, T. Kou, J. B. Varley, S. A. Akhade, S. E. Weitzner, S. E. Baker, E. B. Duoss and Y. Li, *ACS Mater. Lett.*, 2020, **3**, 100–109.

75 C. Peng, G. Luo, J. Zhang, M. Chen, Z. Wang, T. K. Sham, L. Zhang, Y. Li and G. Zheng, *Nat. Commun.*, 2021, **12**, 1580.

76 V. S. S. Mosali, X. Zhang, Y. Liang, L. Li, G. Puxty, M. D. Horne, A. Brajter-Toth, A. M. Bond and J. Zhang, *ChemSusChem*, 2021, **14**, 2924–2934.

77 J. Liu, P. Li, J. Bi, Y. Wang, Q. Zhu, X. Sun, J. Zhang, Z. Liu and B. Han, *Chin. J. Chem.*, 2023, **41**, 1443–1449.

78 Y. Li, Y. Chen, T. Chen, G. Shi, L. Zhu, Y. Sun and M. Yu, *ACS Appl. Mater. Interfaces*, 2023, **15**, 18857–18866.

79 U. Legrand, R. Boudreault and J. L. Meunier, *Electrochim. Acta*, 2019, **318**, 142–150.

80 Y. Gao, Y. Guo, Y. Zou, W. Liu, Y. Luo, B. Liu and C. Zhao, *ACS Appl. Energy Mater.*, 2023, **6**, 1340–1354.

81 H. Cheng, S. Liu, J. Zhang, T. Zhou, N. Zhang, X. S. Zheng, W. Chu, Z. Hu, C. Wu and Y. Xie, *Nano Lett.*, 2020, **20**, 6097–6103.

82 Y. T. J. Chen, Y. Zou, X. Li and J. Jiang, *Mater. Lett.*, 2021, **284**, 128919.

83 D. Yang, Q. Zhu, C. Chen, H. Liu, Z. Liu, Z. Zhao, X. Zhang, S. Liu and B. Han, *Nat. Commun.*, 2019, **10**, 677.

84 A. Saxena, W. Liyanage, J. Masud, S. Kapila and M. Nath, *J. Mater. Chem. A*, 2021, **9**, 7150–7161.

85 J. Wang, X. Zheng, G. Wang, Y. Cao, W. Ding, J. Zhang, H. Wu, J. Ding, H. Hu, X. Han, T. Ma, Y. Deng and W. Hu, *Adv. Mater.*, 2021, **34**, e2106354.

86 H. Zhang, T. Ouyang, J. Li, M. Mu and X. Yin, *Electrochim. Acta*, 2021, **390**, 138766.

87 L. Ding, N. Zhu, Y. Hu, Z. Chen, P. Song, T. Sheng, Z. Wu and Y. Xiong, *Angew. Chem., Int. Ed.*, 2022, **61**, e202209268.

88 J. Duan, T. Liu, Y. Zhao, R. Yang, Y. Zhao, W. Wang, Y. Liu, H. Li, Y. Li and T. Zhai, *Nat. Commun.*, 2022, **13**, 2039.

89 Y. Hao, Y. Sun, H. Wang, J. Xue, J. Ren, A. A. S. Devi, M. Y. Maximov, F. Hu and S. Peng, *Electrochim. Acta*, 2023, **449**, 142213.

90 Y. Deng, Y. Huang, D. Ren, A. D. Handoko, Z. W. Seh, P. Hirunsit and B. S. Yeo, *ACS Appl. Mater. Interfaces*, 2018, **10**, 28572–28581.



91 K. R. Phillips, Y. Katayama, J. Hwang and Y. Shao-Horn, *J. Phys. Chem. Lett.*, 2018, **9**, 4407–4412.

92 R. Elakkiya and G. Maduraiveeran, *Sustain. Energy Fuels.*, 2021, **5**, 6430–6440.

93 C. Guo, Y. Guo, Y. Shi, X. Lan, Y. Wang, Y. Yu and B. Zhang, *Angew. Chem., Int. Ed.*, 2022, **61**, e202205909.

94 P. Li, J. Bi, J. Liu, Q. Zhu, C. Chen, X. Sun, J. Zhang and B. Han, *Nat. Commun.*, 2022, **13**, 1965.

95 A. W. Kahsay, K. B. Ibrahim, M.-C. Tsai, M. K. Birhanu, S. A. Chala, W.-N. Su and B.-J. Hwang, *Catal. Lett.*, 2019, **149**, 860–869.

96 W. He, I. Liberman, I. Rozenberg, R. Ifraimov and I. Hod, *Angew. Chem., Int. Ed.*, 2020, **59**, 8262–8269.

97 J. Li, J. Li, C. Dun, W. Chen, D. Zhang, J. Gu, J. J. Urban and J. W. Ager, *RSC Adv.*, 2021, **11**, 23948–23959.

98 Z. Zhao and G. Lu, *Chem. Sci.*, 2022, **13**, 3880–3887.

99 Z. Zhao and G. Lu, *Adv. Energy Mater.*, 2023, **13**, 2203138.

100 M.-J. Cheng, E. L. Clark, H. H. Pham, A. T. Bell and M. Head-Gordon, *ACS Catal.*, 2016, **6**, 7769–7777.

101 M. Karamad, V. Tripkovic and J. Rossmeisl, *ACS Catal.*, 2014, **4**, 2268–2273.

102 F. Abild-Pedersen, J. Greeley, F. Studt, J. Rossmeisl, T. R. Munter, P. G. Moses, E. Skulason, T. Bligaard and J. K. Norskov, *Phys. Rev. Lett.*, 2007, **99**, 016105.

103 C. Brea and G. Hu, *J. Mater. Chem. A*, 2022, **10**, 10162–10170.

104 Y. Li, G. Shi, T. Chen, L. Zhu, D. Yu, Y. Sun, F. Besenbacher and M. Yu, *Appl. Catal., B*, 2022, **305**, 121080.

105 W. Sun, P. Wang, Y. Jiang, Z. Jiang, R. Long, Z. Chen, P. Song, T. Sheng, Z. Wu and Y. Xiong, *Adv. Mater.*, 2022, **34**, e2207691.

106 H. Wang, X. Bi, Y. Yan, Y. Zhao, Z. Yang, H. Ning and M. Wu, *Adv. Funct. Mater.*, 2023, **33**, 2214846.

107 X. Shao, X. Zhang, Y. Liu, J. Qiao, X.-D. Zhou, N. Xu, J. L. Malcombe, J. Yi and J. Zhang, *J. Mater. Chem. A*, 2021, **9**, 2526–2559.

108 S. Kumaravel, K. Karthick, P. Thiruvengetam, J. M. Johny, S. S. Sankar and S. Kundu, *Inorg. Chem.*, 2020, **59**, 11129–11141.

