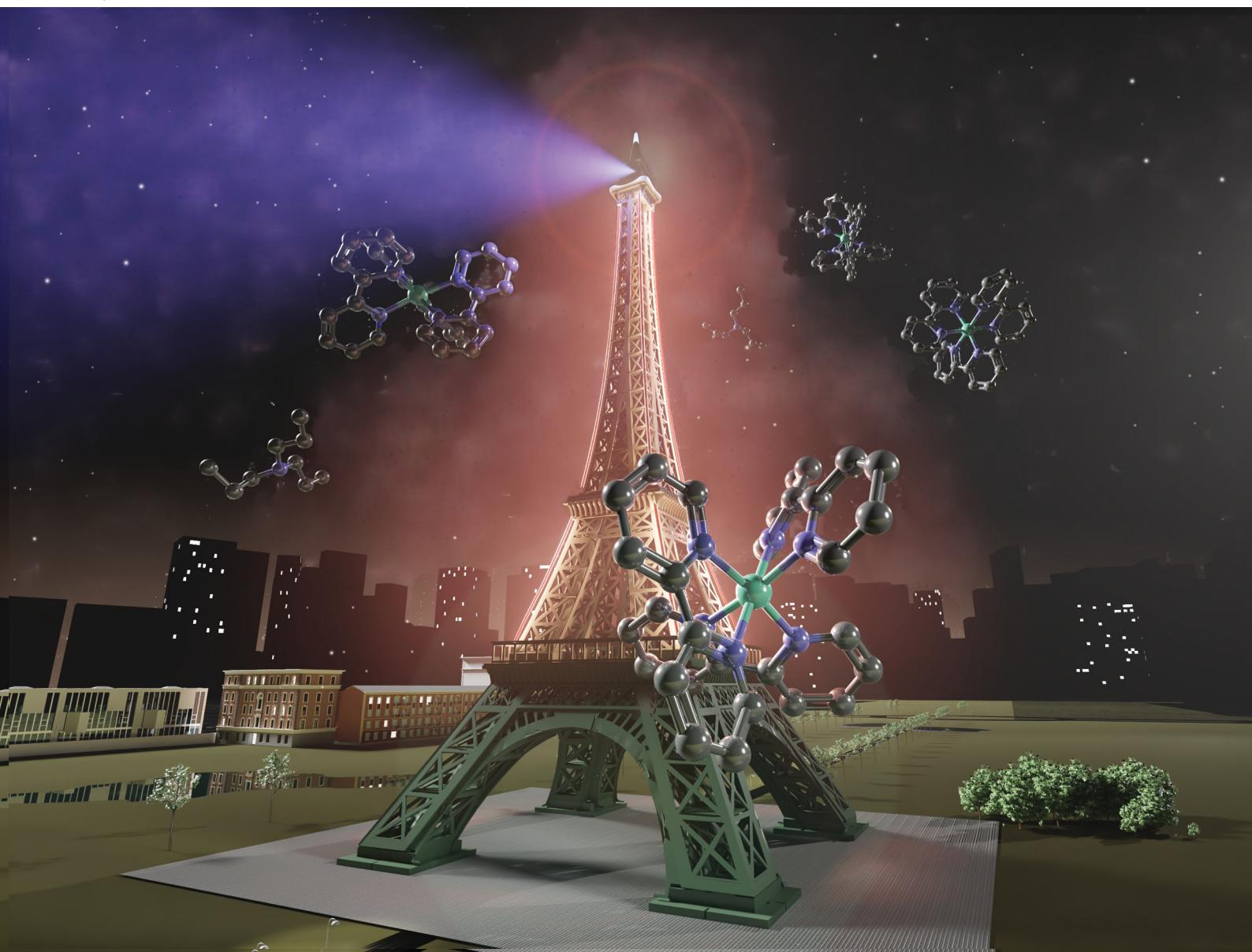


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Local reactivity of metal–insulator–semiconductor photoanodes imaged by photoinduced electrochemiluminescence microscopy[†]

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Localized photoinduced electrochemiluminescence (PECL) is studied on photoanodes composed of Ir microbands deposited on n-Si/SiO_x. We demonstrate that PECL microscopy precisely imaged the hole-driven heterogeneous photoelectrochemical reactivity. The method is promising for elucidating the local activity of photoelectrodes that are employed in solar energy conversion.

Photoelectrochemistry of illuminated semiconductors is a field of research that deals with the charge transfer of photogenerated charge carriers (electrons and holes) at the solid–liquid interface.^{1,2} It has promising applications in the field of energy conversion as the basis for artificial photosynthesis, in which solar energy is converted into H₂,^{3,4} C1, or C2^{5–7} products using water and/or CO₂ as reactants. In these systems, semiconductor photoelectrodes, often coated with protection and/or catalytic layers are immersed in a liquid electrolyte.^{8–10} Upon light absorption, photogenerated charge carriers diffusing within the semiconductor, are collected at the catalyst and trigger redox reactions in the liquid phase. Photoelectrochemical characterization is mainly done by recording the current density (*j*) as a function of another parameter, for instance, the potential (*E*)/*j* (i.e., voltammetry) or time (*t*)/*j* (i.e., chronoamperometry) measurements, in which the *j* value corresponds to the faradaic charge transfer rate at the photoelectrode surface. Although these electroanalytical methods are straightforward for performance assessment, the *j* value only corresponds to the overall charge transfer rate, averaged over all the electrode

surface, and does not allow for spatially-resolved determination of photoelectrochemical activity. However, probing the local activity is essential for elucidating the operation of photoelectrodes that are made by combining several materials, usually a semiconductor absorber, a protection layer, and a catalytic coating⁷ (in the form of thin films^{11,12} or particles^{10,13–16} dispersed over the surface). Due to the complexity of these interfaces, the photoelectrochemical reaction is not expected to occur at the same rate over the whole surface. Visualizing local photoelectrochemical activity remains a challenge, which has been investigated by several groups¹⁷ using different methods such as laser scanning techniques,¹⁸ scanning electrochemical microscope variants,^{19–24} or a combination of both.²⁵

Electrochemiluminescence (ECL) is a light-emitting process initiated at an electrode surface.²⁶ The light originates from a luminophore molecule that is brought to its excited state by a highly exergonic electron-transfer reaction and relaxes to its ground state by emitting a photon.^{27,28} This phenomenon is widely used for medical diagnosis applications and immunoassays,^{29–36} with the ECL model system comprising tris(bipyridine)ruthenium(II) complex ([Ru(bpy)₃]²⁺) and tri-*n*-propylamine (TPA) as the luminophore and the co-reactant, respectively. Since ECL offers an optical readout, it has evolved progressively into a microscopy technique.^{37–42} Single entities such as cells, bacteria, and organelles have been imaged by ECL with remarkable resolution.^{39,43–49} Notably, ECL has also been employed for visualizing local electron transfer and catalytic activity of single nano-objects and particles.^{50–55} Photoinduced ECL (PECL) combines semiconductor photoelectrochemistry and ECL.⁵⁶ In this process, a light input (λ_{exc}) is electrochemically converted and generates a light output (λ_{PECL}) through an ECL reaction. Depending on the choice of the electrode semiconductor material and the ECL system implied, PECL can allow either a downconversion ($\lambda_{\text{PECL}} < \lambda_{\text{exc}}$)⁵⁷ or an upconversion ($\lambda_{\text{PECL}} > \lambda_{\text{exc}}$) emission (Table S1, ESI[†]).^{56,58–63} So far, PECL microscopy has been reported in two studies, and used to probe upconversion PECL at Au nanoparticles localized on Si

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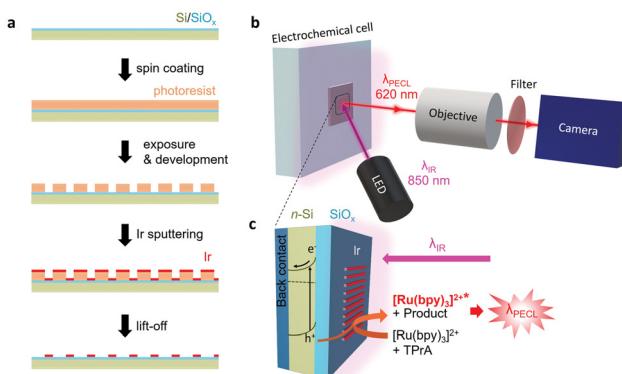
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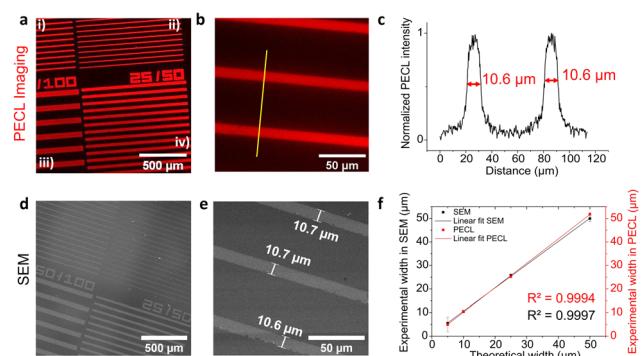


nanopillars⁶⁴ or TiO₂.⁶² In these reports, PECL was triggered through the oxidation of luminol *via* photogenerated holes.

Herein, we report the detailed PECL microscopy study of inhomogeneous metal-insulator-semiconductor (MIS) photoanodes using the model $[\text{Ru}(\text{bpy})_3]^{2+}/\text{TPrA}$ system (Fig. 1). Such MIS-based architectures are often employed for manufacturing efficient and stable Si-based water-splitting photoanodes.^{65–68} In particular, MIS photoanodes based on Si and Ir have been previously employed for photoelectrochemical water splitting^{66,68–70} and it is known that n-Si/SiO_x/Ir photoanodes, in their homogeneous (*i.e.*, non-patterned) form exhibit exceptional PECL stability (Table S1, ESI†).⁶³ Here, inhomogeneous n-Si/SiO_x/Ir photoanodes are prepared by sputtering Ir microbands with a defined pattern. The operation of upconversion PECL and observation of the produced emission patterns allows resolving the solid/liquid photo-electrogenerated hole transfer at the local scale.

The MIS photoanodes were fabricated by chemical oxidation, photolithography and magnetron sputtering (Fig. S1, ESI†). First, Si/SiO_x was prepared by oxidizing a moderately doped photoactive n-Si surface to create a $\approx 1.5 \text{ nm}$ -thick tunnel SiO_x layer, as described previously.^{67,71} Prior to the PECL experiments, a 2 nm-thick Ir pattern (Fig. S2, ESI†) was deposited on Si/SiO_x using a lift-off process, described in Fig. 1a (see details in ESI†). Four types of Ir patterns with various width/pitch (50 μm /100 μm , 25 μm /50 μm , 10 μm /50 μm , and 5 μm /50 μm) were designed and produced. It is worth mentioning that the Si/SiO_x and the Si/SiO_x/Ir junctions were characterized by X-ray photoelectron spectroscopy and atomic force microscopy in a previous paper.⁶³ For the PECL microscopy experiments, the photoanode was excited by side-illumination with a near-IR LED ($\lambda_{\text{exc}} = 850 \text{ nm}$) to photogenerate charge carriers, as shown in Fig. 1b and c. The localized PECL emission was imaged with a microscope equipped with an IR filter (Fig. S3, ESI†).

As shown in Fig. 2, microscopy images of PECL could be reliably recorded on n-Si/SiO_x/Ir under near-IR illumination at



0.8 V (all potential are here referred *vs.* Ag/AgCl) in the electrolytic solution containing both $[\text{Ru}(\text{bpy})_3]^{2+}$ and TPrA. Fig. 2a shows that the red PECL light is homogeneously emitted over the Ir microbands, for all the patterns. At higher magnification (Fig. 2b), the PECL sharply delimits the contour of the Ir surface area allowing an excellent definition of the patterns, as further demonstrated by the intensity profile in Fig. 2c (*vide infra*). The bands made of Ir thin film are easily viewed by PECL because Ir acts as a collector for photogenerated charges and presents good charge transfer properties, leading to the efficient oxidation of both $[\text{Ru}(\text{bpy})_3]^{2+}$ and TPrA producing light locally.⁶³

In contrast, Si surface is passivated in the aqueous solution at the anodic potentials required for ECL generation. In other words, PECL is preferentially emitted at the Ir patterns (Fig. 2a and b). The 2 nm-thick Ir pattern was not observable with the naked eye or under a wide-field microscope, and therefore, scanning electron microscopy (SEM) was required to characterize the thin film patterns. Fig. 2d and e presents the SEM pictures of the Ir microbands that allow for a precise measurements of their dimensions. Then, we compared their dimensions measured by SEM and by PECL. For PECL microscopy, we extracted the PECL intensity profiles along a line perpendicular to the long band axis. More precisely, their full-width at half-maximum (FWHM) values were compared with the width measured by SEM for all the patterns: 10 μm /50 μm in Fig. 2c and, 50 μm /100 μm , 25 μm /50 μm , and 5 μm /50 μm in Fig. S4c, f and i, ESI†). In Fig. 2f, the PECL and SEM width measurements were plotted as a function of the theoretical width. The linear fits of both measurements show a slope close to 1 ($R^2 = 0.9997$ and $R^2 = 0.9994$ for the SEM and the PECL, respectively). This reveals the accuracy of the Ir pattern manufacturing with the lift-off process on one hand, and, on the other hand, that the PECL is restricted to the conductive patterns. We selected a

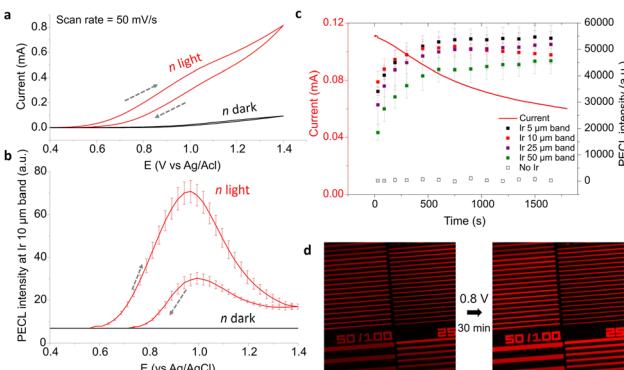


Fig. 3 (a) Voltammetric measurement of n-Si/SiO_x patterned with Ir microbands in the dark (black curve) and under LED illumination (red curve). The dashed grey arrows show the sweep direction. (b) Corresponding average PECL intensity of the 10 μm/50 μm Ir microbands under IR illumination (red curve) and in the dark (black curve). (c) Evolution of the current (line) and the PECL of each Ir band (squares) as a function of time at $E_{app} = 0.8$ V under LED light. (d) PECL imaging before (left) and after (right) a potentiostatic measurement. Same solution as in Fig. 2.

high concentration of TPrA (*i.e.* 0.1 M) for the PECL imaging experiments because (i) it generates a strong PECL signal and (ii) it confines the PECL-emitting layer at the electrode surface. Indeed, in the present experimental conditions, the size of the PECL reaction layer is typically ≈ 200 nm so much smaller than the dimensions of the bands.⁷² Control experiments, presented in Fig. S5 (ESI[†]), were performed under illumination at open-circuit conditions to ensure that the observed pattern was not caused by reflection of the IR light. Furthermore, Fig. S5c (ESI[†]) shows that polarization of the electrode at 0.8 V in the dark did not lead to PECL emission, confirming the necessity of photo-carrier generation in the PECL process.

We further investigated the photoelectrochemistry of the n-Si/SiO_x/Ir electrode. A cyclic voltammogram (CV) was recorded in the same electrolyte (5 mM [Ru(bpy)₃]²⁺, 0.1 M TPrA, 0.1 M PBS at pH = 7.4) under IR illumination at $\lambda_{exc} = 850$ nm with a power density (P_{LED}) of 6.6 mW cm⁻² (red curve) and in the dark (black curve, Fig. 3a). The PECL images were captured with the microscope at a speed of 2 frames per second (fps). In Fig. 3b, the corresponding average PECL intensity was plotted against the applied potential for the 10 μm/50 μm microbands. This points out that the local PECL emission was only initiated from +0.57 V. These measurements can be compared with that of an analogous but non-photoactive p⁺⁺-Si/SiO_x/Ir MIS anode prepared on a highly-doped, degenerate p⁺⁺-Si surface and studied in the dark. Indeed, ECL microscopy on these anodes revealed that they also promote local ECL at the Ir/electrolyte interface, however, at higher potentials than that required for their illuminated photoactive counterpart (n-Si/SiO_x/Ir). This shift of ECL onset potential can be clearly observed in Fig. S6a (ESI[†]) where the CV obtained for p⁺⁺-Si/SiO_x/Ir and n-Si/SiO_x/Ir under different illumination conditions are plotted together. This shift of onset potential (*here* ≈ 0.5 V) is characteristic of a conventional photoanode behavior and caused by the generation of a photovoltage, this was already observed (*to a lower extent*) on non-patterned n-Si/SiO_x/Ir photoanodes.^{63,73}

We then studied the evolution of PECL on the illuminated n-Si/SiO_x/Ir over time by potentiostatic measurement at 0.8 V for 30 min while capturing microscopy pictures. The PECL intensity averages were measured on the pictures for the 50 μm/100 μm, 25 μm/50 μm, 10 μm/50 μm, and 5 μm/50 μm microbands. On Fig. 3c, it can be observed that the photocurrent decreases by $\sim 45\%$ after 30 min, but, meanwhile, the PECL intensities increased and reached a plateau after about 10 min. The pictures at the beginning and at the end of the potentiostatic measurement, (recorded with the same conditions of the CCD camera with $t_{expo} = 0.5$ s) complete this observation with a PECL that appears more intense at the end of the experiment (Fig. 3d). This behavior could be explained by the anodic passivation of the Si/SiO_x surface that inhibits the flow of residual current onto the Ir-free area and the surface oxidation of Ir that could influence the ECL process. However, the pattern remains identical after 30 min, showing no obvious damage of the patterns. In addition to the PECL on n-Si/SiO_x/Ir, the ECL intensity on the p⁺⁺-Si/SiO_x/Ir electrode was also recorded (*in the dark at 1.2 V, Fig. S6b and S7, ESI[†]*), confirming our observations.

To conclude, we have performed a PECL microscopy study of MIS n-Si/SiO_x/Ir microbands. This junction has been considered so far as the most stable interface for manufacturing PECL photoanodes (Table S1, ESI[†])⁶³ and has implications in the field of photoelectrochemical solar energy conversion.^{66,68–70} Inhomogeneous photoactive surfaces comprising ultrathin Ir microbands were prepared on Si/SiO_x and studied under an optical microscope by PECL under IR irradiation ($\lambda_{exc} = 850$ nm). The emission patterns ($\lambda_{PECL} = 635$ nm) were intense, stable, and presented geometrical features identical to that of the Ir microbands. These results clearly revealed, in the form of a snapshot, that photoelectrochemical charge transfer occurs at the surface Ir microbands and is inhibited at the SiO_x surface. We anticipate that PECL microscopy can be used as a novel approach for understanding the complexity of inhomogeneous solid/liquid junctions and could help the further development in photoelectrochemical solar energy conversion.

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Conflicts of interest

There are no conflicts to declare.

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