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Colorimetric response in polydiacetylene at the single domain level using hyperspectral microscopy†

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The structural variance of polydiacetylene (PDA) at the nanoscale level, even under the same fabrication conditions, is one of the origins of its poor reproducibility in chemo/biosensing. In this work, we present a spatial map of such structural distributions within a single crystal by taking advantage of the recent development of hyperspectral microscopy at visible wavelengths. Hyperspectral microscopy provides the distribution of absorption spectra at the spatial resolution of standard optical microscopy. By tracking the blue-to-red transition via this technique, we found that heat or pH stimulation leaves a unique pattern in the transition pathways.

Polydiacetylenes (PDAs) are a class of mechanochromic polymers that present a blue-to-red color transition and emit fluorescence when triggered by various stimuli such as heat,¹ pH,² mechanical stress,³ biomolecules,⁴ and so on. Being encouraged by their customizable chemical structure and facile fabrication method,⁵ much effort has been devoted to developing PDA sensors⁶ involving biomolecular detection *via* receptor–ligand interactions,⁷ food deterioration *via* pH monitoring,⁸ and volatile organic compound identification.⁹ However, some challenges remain for bringing these technologies forward to real-life applications.¹⁰

One of them is the unwanted variation of PDA structures, where the fabrication procedures seem to yield a mixture of PDAs with slightly different structures. Such distribution can vary from batch to batch, worsening the reproducibility of chemo/biosensing.¹¹ Many factors, such as the monomer distance during polymerization,¹² the alkyl spacer properties,¹³ and the lipid composition and size,¹⁴ lead to subtle differences in the PDA structure, which ultimately affect the mechanochromic properties. Raman spectroscopy, infrared spectroscopy (IR), X-ray diffraction, and nuclear magnetic resonance (NMR) have been utilized to decipher the PDA structure during polymerization and the blue-to-red transition.^{15–19} However, they

are all based on ensemble measurements, and fail to capture the distribution of different structures within a sample.

In this work, we introduce hyperspectral microscopy that operates in the visible wavelength range to visualize the distribution of different structures at the sub-crystal level. Hyperspectral microscopy provides an absorption spectral map with a spatial resolution that is comparable to standard optical microscopy. Hyperspectral imaging dates back to the 1970s when NASA mapped infrared (IR) spectra for remote geology sensing²⁰ and mineral detection.²¹ Since its first commercialization in the late 1980s, recent improvements in the computing technology and the detector have significantly broadened its applications towards medical diagnosis,²² chemical identification,²³ quality monitoring,²⁴ agriculture,²⁵ astronomy²⁶ and geological studies.²⁷ By applying this novel technique to PDA research, we will show that both blue and red PDA films contain various structures even within a single crystal, and that each structure responds differently to stimuli. Tracking the transformation between these blue and red structures yields a unique pattern of the transition pathways.

Blue PDA is made of microdomains with at least 10 different structures. PDA made of 10,12-tricosadiynoic acid (TRCDA) assembled on glass coverslips *via* drop-casting of the monomer solution in chloroform, followed by UV irradiation (see Materials and methods (ESI†) for details). Bright-field microscopy imaging captured the PDA crystals, which look uniformly blue (Fig. 1a). However, the mapping of their absorption spectra *via* hyperspectral microscopy revealed that the PDA sample consists of many different “blues”, where their spectra are slightly different from each other (Fig. 1c; note that the top 10 spectra were color-coded and their spatial distribution is shown in Fig. 1b).

The spectral angle mapper method (see Material and methods (ESI†) for the details) that we used for classification does not distinguish the spectra that can be superposed by the multiplication of their amplitudes. This avoids categorizing PDA by thickness. This suggests that each blue spectrum that we obtained (B1–B10) corresponds to a different structure of PDA. This is because each PDA absorption spectrum reflects the

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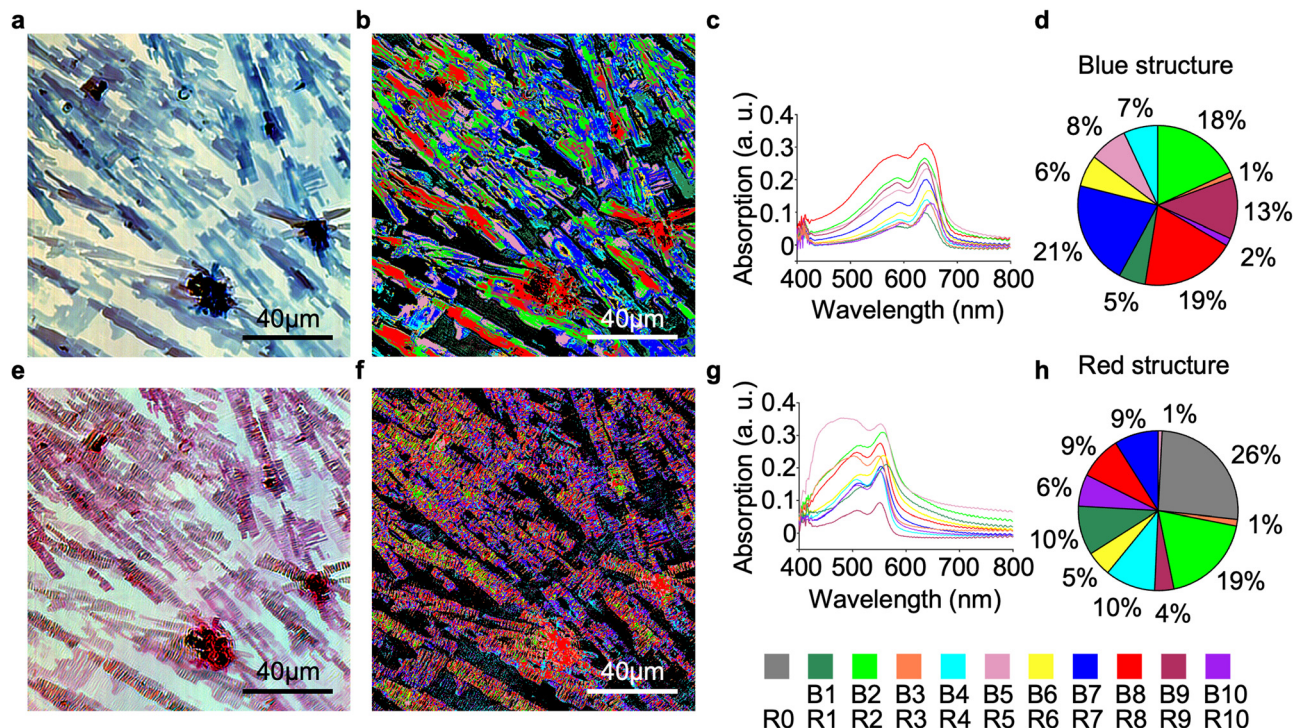


Fig. 1 Structural mapping of blue and red PDA crystals. (a and e) Optical image of blue and red PDA, respectively. (b and f) Spectral map of blue and red PDA, respectively. (c and g) Absorption spectra of blue and red PDA, respectively. (d and h) Structure distribution of blue and red PDA, respectively. Details such as the error bars are shown in Fig. S1 (ESI[†]). Note that the spectra are color-coded, and the same color code is used for (b–d) and (f–h).

electronic structure at the conjugated backbone, which is influenced by the polymer length and its architecture. This inhomogeneity originates from the lipidic monomer self-assembly process. Lipid self-assembly produces structures that are energetically favorable under the experimental conditions, yet in reality it often yields a distribution of various architectures, such as those with different packing arrangements, lipid tail angle relative to the monolayer leaflet, *etc.* To obtain statistics, we analyzed 11 samples. Their average showed that the spectra B7, B8, and B2 correspond to 21%, 19%, and 18% of the entire PDA crystals, respectively, being the top three major structures in the blue PDA (Fig. 1d). These structures seem to form microdomains in the crystals (Fig. 1b), where some PDA crystals contain more than two structures. This implies that the structure at the nanoscale is not uniform, even within a single crystal.

Red PDA also consists of at least 10 different structures, but the domain size is smaller than that of blue PDA. Next, we heated these blue PDA samples at 240 °C for 1 min on the microscope (see Materials and methods (ESI[†])) to convert them into red PDAs before characterizing them *via* hyperspectral microscopy. The sample appeared to be red in a relatively uniform manner (Fig. 1e), yet the spectral mapping visualized that it contains at least ten different “reds” (Fig. 1f and g). Interestingly, their domain size became smaller on average (Table S1, ESI[†]), which implies an increase in the entropy, as has previously been reported.^{28,29} As a result, a single crystal accommodates many sub-micrometer domains that have different structures. Averaging

the results from 11 samples shows that there are also some major red structures (Fig. 1h, *e.g.*, R2, R1, and R4 of 19%, 10%, and 10%, respectively). The 26% grey portion represents glass areas that are newly created during the blue-to-red transition because of the shrinkage of the crystals. This implies that the PDA contracted upon heating. A similar reduction of the area during the blue-to-red transition has been reported before.³⁰

Hyperspectral dark-field microscopy has previously been applied to peptide-PDA films made of 10,12-pentacosadiynoic acid to confirm its homogeneity.³¹ Nevertheless, observed inhomogeneity and microdomains in our samples have been also previously reported with other types of PDA. Gaber's group observed that PDA vesicles made of 1,2-bis(tricoso-10,12-diynoyl)-*sn*-glycero-3-phosphocholine (DC₈₉PC) yielded different sizes, as characterized using size exclusion chromatography, and that each of them presented unique properties.³² The addition of a surfactant has also been shown to alter the PDA vesicle size, where the smaller vesicles were more sensitive.³³ Korostylev and co-workers reported that optimization of the distance and the spatial arrangement of diacetylene monomers are critical for efficient polymerization, and thus affect the final polymer structures and their functions.³⁴ Jelinek's group studied TRCDA Langmuir films using a Brewster angle microscope and has found that monolayer and multilayer domains were mixed in single samples.³⁵ Atomic force microscopy (AFM)^{36,37} and transmission electron microscopy (TEM)³⁸ have also been used to visualize microdomains within PDA crystals. These different structures mixed in PDA samples were suspected to



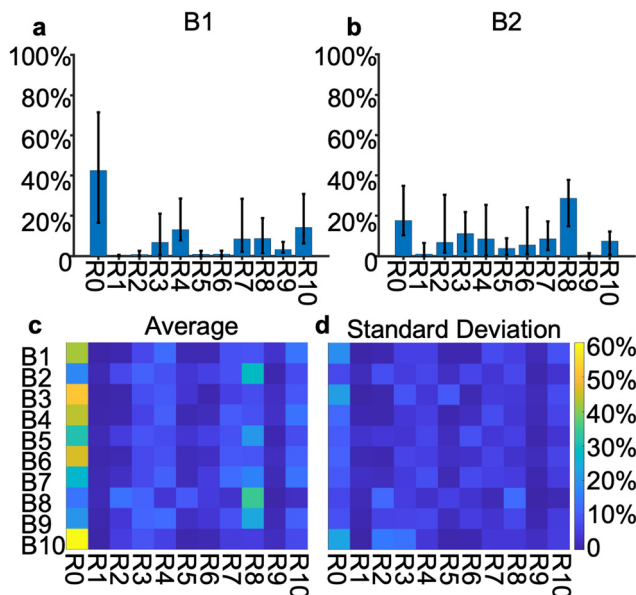


Fig. 2 Unique pattern of the blue-to-red transition pathways. (a and b) Pixel fraction of B1 and B2 that transformed into different red structures (R0–R10); and (c) the average and (d) the standard deviation of these transitions from B1–B10 to R0–R10.

be one of the origins of the poor reproducibility in PDA bio/chemosensing.

There is a unique pattern for the structural transformation from blue to red PDA. To decipher the transformation behavior from blue to red stimulated by heat, we studied, as a percentage, how much of B1 transformed into R0, R1, ... and R10, respectively (Fig. 2a). On average, 43% of the pixels that had B1 spectra were transformed into R0 (glass area) after heating, whereas B1 → R1, B1 → R2, B1 → R5, and B1 → R6 were negligible pathways because their portions were less than 3%. Similarly, we analyzed the transition pathways from B2 (Fig. 2b), and up to B10 (Fig. S2, ESI†). All of the results are summarized in Fig. 2c and d for average and standard deviation data, respectively. R8 seems to be the preferred-endpoint red structure for many blue structures, besides R0 (glass area). By contrast, R1, R5, R6, and R9 seem to be unpreferred red structures as their transition portions are small for most of the blue structures. Overall, the standard deviation is not necessarily small, yet for some of the major pathways such as B8 → R8 the value (35%) is larger compared with the standard deviation (13%), suggesting that the blue-to-red transition induced by heat takes place with a certain pattern. Note that this pattern is visible regardless of the error in the spatial superposition between the blue and the red polymers due to shrinkage of the crystals.

The structural change during the blue to red transition has been extensively studied before. ¹³C NMR indicated the twist of the backbone, where the deplanarization of the alkyl side chains was detected during the blue-to-red transition.³⁹ NMR and Raman spectroscopy, combined with theoretical calculations, have shown that the PDAs made of poly(1,6-bis(diphenylamino)-2,4-hexadiyne) (poly-THD) had twisted, slightly-twisted, and flat (planar) side chains

for the red, intermediate, and blue forms, respectively, where the strongly-twisted red polymer was the most stable and the major red structure.⁴⁰ By contrast, AFM and Fourier-transform infrared spectroscopy (FTIR) indicated an increase in the molecular-scale ordering and the near-perfect hexagonal packing of alkyl side chains as a response to the heat stimulus.⁴¹ As such, different blue structures have been suggested in the literature and a variety of methods have detected that each blue form has its distinct response against stimuli. Although we do not know what the exact structures of B1–B10 and R1–R10 are, their structural differences probably originate from a slight shift in the monomer distance, the carbon chain angle relative to the substrate or to the backbone, or different lipid phases such as lamellar and micelle architectures and their mixtures. Each form of PDA potentially has a different sensitivity to heat and transfers into different red structures. Fig. 2c shows the first comprehensive map of such transitions from various blue to various red structure for PDA made using TRCDA stimulated by heat. To study whether this transition map is unique in each stimulus, next we selected thin blue PDA areas, stimulated them using heat and pH, and obtained their blue-to-red transition patterns *via* the same analytical procedure (Fig. S3 and S4, ESI†). Note that new sets of 10 blue and 10 red spectra were used in this analysis; therefore, B1, for example, in Fig. 1 and 3 are not identical. The results show that each stimulus (heat or pH) leaves a unique (different) transition pattern (Fig. 3). This indicates the potential for using such a map to identify different stimuli. These transition maps were reproducible in crystal structures, but less so in amorphous PDAs (Fig. S5, ESI†).

We visualized the distribution of the microdomains from different blue and red structures by introducing hyperspectral microscopy that operates at visible wavelengths. This was enabled

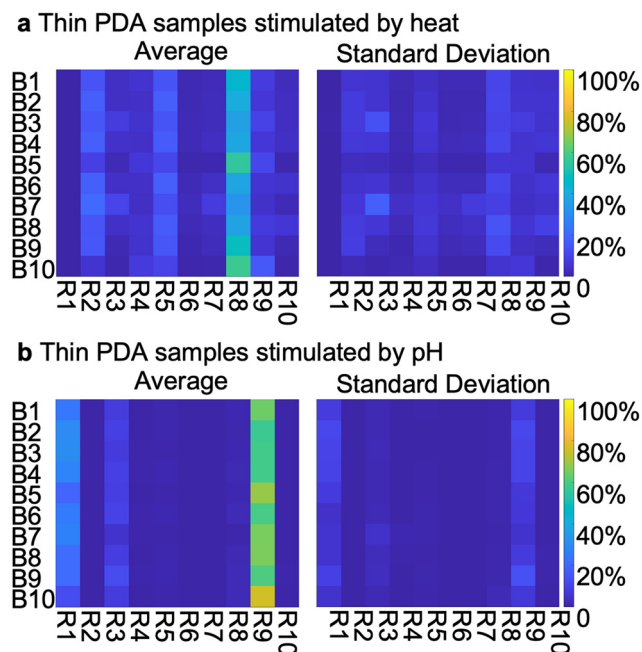


Fig. 3 Blue-to-red transition map of thin PDAs stimulated using (a) heat and (b) pH. Note that a new set of B1–B10 were used in this analysis compared with those used in Fig. 1 and 2.



due to the spectral mapping ability of the technique with a spatial resolution that goes below the single crystal size, in contrast to conventional ensemble measurements such as UV-Vis, IR, Raman spectroscopy, NMR, X-ray diffraction, and other imaging methods that have the spatial resolution but not the spectral information such as AFM and TEM.⁴² Other spectral mapping based on scanning microscopy methods, such as infrared scattering scanning near-field optical microscopy (IR-s-SNOM)^{43,44} and tip-enhanced Raman spectroscopy (TERS), may be an alternative tool.⁴⁵ We found that each blue structure undergoes a unique path to diverse types of red structure, where a pattern, *e.g.*, a certain blue structure prefers to transform into a certain red structure, was detected. The result implies that the stimulus (heat, in this work) leaves a fingerprint to the PDA blue-to-red transition pathways. Deconvoluting the spectra at the sub-crystal domains and tracking their blue-to-red transitions separately instead of observing their average may help with tackling the sample inhomogeneity issue and the related reproducibility problem in PDA sensing. In addition, the hyperspectral microscopy that is used in this work, which can be also operated in fluorescence mode, can be applied to other (mechano)chromic materials in the future.⁴⁶

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Conflicts of interest

There are no conflicts to declare.

References

- 1 C. Girard-Reydet, R. D. Ortuso, M. Tsemperouli and K. Sugihara, *J. Phys. Chem. B*, 2016, **120**, 3511–3515.
- 2 S. J. Kew and E. A. H. Hall, *Anal. Chem.*, 2006, **78**, 2231–2238.
- 3 L. Juhasz, R. D. Ortuso and K. Sugihara, *Nano Lett.*, 2021, **21**, 543–549.
- 4 J. Zhao and K. Sugihara, *J. Phys. Chem. B*, 2021, **125**, 12206–12213.
- 5 D. J. Ahn, S. Lee and J.-M. Kim, *Adv. Funct. Mater.*, 2009, **19**, 1483–1496.
- 6 B. Yoon, S. Lee and J.-M. Kim, *Chem. Soc. Rev.*, 2009, **38**, 1958–1968.
- 7 D. H. Charych, J. O. Nagy, W. Spevak and M. D. Bednarski, *Science*, 1993, **261**, 585–588.
- 8 M. Weston, R. P. Kuchel, M. Ciftci, C. Boyer and R. Chandrawati, *J. Colloid Interface Sci.*, 2020, **572**, 31–38.
- 9 S. Dolai, S. K. Bhunia, S. S. Beglaryan, S. Kolusheva, L. Zeiri and R. Jelinek, *ACS Appl. Mater. Interfaces*, 2017, **9**, 2891–2898.
- 10 M. Weston, A. D. Tjandra and R. Chandrawati, *Polym. Chem.*, 2020, **11**, 166–183.
- 11 Y. K. Jung, T. W. Kim, H. G. Park and H. T. Soh, *Adv. Funct. Mater.*, 2010, **20**, 3092–3097.
- 12 S. Okada, S. Peng, W. Spevak and D. Charych, *Acc. Chem. Res.*, 1998, **31**, 229–239.
- 13 R. Potai, K. Faisadcha, R. Traiphol and N. Traiphol, *Colloids Surf., A*, 2018, **555**, 27–36.
- 14 A. Nopwinyuwong, T. Kitaoka, W. Boonsupthip, C. Pechyen and P. Suppakul, *Appl. Surf. Sci.*, 2014, **314**, 426–432.
- 15 D. J. Sandman and Y. J. Chen, *Polymer*, 1989, **30**, 1027–1031.
- 16 D. N. Batchelder, N. J. Poole and D. Bloor, *Chem. Phys. Lett.*, 1981, **81**, 560–564.
- 17 D. N. Batchelder and D. Bloor, *J. Phys. C: Solid State Phys.*, 1982, **15**, 3005.
- 18 E. Armando Zaragoza-Contreras, C. A. Hernández-Escobar and I. Rodríguez-Sánchez, *Spectrochim. Acta, Part A*, 2022, **267**, 120506.
- 19 R. D. Ortuso, N. Ricardi, T. Bürgi, T. A. Wesolowski and K. Sugihara, *Spectrochim. Acta, Part A*, 2019, **219**, 23–32.
- 20 S. F. Alexander, H. Goetz, G. E. Vane Jerry Solomon and B. N. Rock, *Science*, 1985, **228**, 1147–1153.
- 21 G. A. Roth, S. Tahiliani, N. M. Neu-Baker and S. A. Brenner, *Wiley Interdiscip. Rev.: Nanomed. Nanobiotechnol.*, 2015, **7**, 565–579.
- 22 G. Lu and B. Fei, *J. Biomed. Opt.*, 2014, **19**, 010901.
- 23 V. Farley, M. Chamberland, P. Lagueux, A. Vallières, A. Villemaire and J. Giroux, *SPIE Proc.*, 2007, **6739**, 673918.
- 24 H. Huang, L. Liu and M. O. Ngadi, *Sensors*, 2014, **14**, 7248–7276.
- 25 L. M. Dale, A. Thewis, C. Boudry, I. Rotar, P. Dardenne, V. Baeten and J. A. F. Pierna, *Appl. Spectrosc. Rev.*, 2013, **48**, 142–159.
- 26 M. Pisani and M. Zucco, *SPIE-Int. Soc. Opt. Eng.*, 2018, **10677**, 1067706.
- 27 S. Peyghambari and Y. Zhang, *J. Appl. Remote Sens.*, 2021, **15**, 31501.
- 28 S. J. Kew and E. A. H. Hall, *J. Mater. Chem.*, 2006, **16**, 2039–2047.
- 29 J. Lee, S. Seo and J. Kim, *ACS Appl. Mater. Interfaces*, 2018, **10**, 3164–3169.
- 30 J. Nuck and K. Sugihara, *Macromolecules*, 2020, **53**, 6469–6475.
- 31 Y. Kuang, Z.-F. Yao, S. Lim, C. Ngo, H. Ann and M. Ardoña, *Steric and Hydrophobic Control Over Sequence-Defined Peptide Templates to Modulate the Properties of Polydiacetylene Nanomaterials*, ChemRxiv, 2022, DOI: [10.26434/chemrxiv-2022-7zf4p](https://doi.org/10.26434/chemrxiv-2022-7zf4p).
- 32 B. M. Peek, J. H. Callahan, K. Nambodiri, A. Singh and B. P. Gaber, *Macromolecules*, 1994, **27**, 292–297.
- 33 A. Nopwinyuwong, T. Kitaoka, W. Boonsupthip, C. Pechyen and P. Suppakul, *Appl. Surf. Sci.*, 2014, **314**, 426–432.
- 34 R. v Vizgert and A. P. Korostylev, *Polym. Sci. U.S.S.R.*, 1989, **31**, 380–387.
- 35 F. Gaboriaud, R. Volinsky, A. Berman and R. Jelinek, *J. Colloid Interface Sci.*, 2005, **287**, 191–197.
- 36 R. W. Carpick, D. Y. Sasaki and A. R. Burns, *Tribol. Lett.*, 1999, **7**, 79–85.
- 37 A. Lio, A. Reichert, J. O. Nagy, M. Salmeron and D. H. Charych, *J. Vac. Sci. Technol., B*, 1996, **14**, 1481–1485.
- 38 Q. Ye, G. Zou, X. You, X. Yu and Q. Zhang, *Mater. Lett.*, 2008, **62**, 4025–4027.
- 39 H. Tanaka, M. Thakur, M. A. Gomez and A. E. Tonelli, *Polymer*, 1991, **32**, 1834–1840.
- 40 J.-S. Filhol, J. Deschamps, S. G. Dutremez, B. Boury, T. Barisien, L. Legrand and M. Schott, *J. Am. Chem. Soc.*, 2009, **131**, 6976–6988.
- 41 A. Lio, A. Reichert, D. J. Ahn, J. O. Nagy, M. Salmeron and D. H. Charych, *Langmuir*, 1997, **13**, 6524–6532.
- 42 A. Szerlauth, E. Balog, D. Takács, S. Sáringner, G. Varga, G. Schusztzer and I. Szilágyi, *Colloid Interface Sci. Commun.*, 2022, **46**, 100564.
- 43 Q. Zhu, R. Zhou, J. Liu, J. Sun and Q. Wang, *Nanomaterials*, 2021, **11**, 1353.
- 44 F. Zhong, H. Wang, Z. Wang, Y. Wang, T. He, P. Wu, M. Peng, H. Wang, T. Xu, F. Wang, P. Wang, J. Miao and W. Hu, *Nano Res.*, 2021, **14**, 1840–1862.
- 45 K. Jajcevic, A. M. Sequeira, J. Kalbacova, D. R. T. Zahn and K. Sugihara, *Nanoscale*, 2021, **13**, 6927–6933.
- 46 H. Traeger, Y. Sagara, D. J. Kiebalá, S. Schrettl and C. Weder, *Angew. Chem., Int. Ed.*, 2021, **60**, 16191–16199.

