

## PAPER

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[View Journal](#) | [View Issue](#)Cite this: *J. Mater. Chem. A*, 2022, **10**, 21738Topotactic  $\text{Bl}_3$ -assisted borodization: synthesis and electrocatalysis applications of transition metal borides†Katherine E. Woo,<sup>‡ab</sup> Seongyoung Kong,<sup>ab</sup> Wei Chen,<sup>‡c</sup> Tsz Hin Chang,<sup>d</sup> Gayatri Viswanathan,<sup>ab</sup> Aida M. Díez,<sup>e</sup> Viviana Sousa,<sup>ab</sup> Yuri V. Kolen'ko,<sup>e</sup> Oleg I. Lebedev,<sup>f</sup> Marta Costa Figueiredo<sup>bc</sup> and Kirill Kovnir<sup>ab</sup>\*

A facile and universal route for synthesizing transition metal borides has been developed by reaction of boron triiodide ( $\text{Bl}_3$ ) with elemental transition metals. This method employs relatively low synthesis temperatures to afford single-phase samples of various binary and ternary metal borides, such as  $\text{Fe}_2\text{B}$ ,  $\text{Co}_2\text{B}$ ,  $\text{Ni}_3\text{B}$ ,  $\text{TiB}_2$ ,  $\text{VB}_2$ ,  $\text{CrB}_2$ , and  $\text{Ni}_2\text{CoB}$ . This synthesis protocol can be utilized for the topotactic transformation of metal shapes into their respective borides, as exemplified by transformation of Ni foam to  $\text{Ni}_3\text{B}$  foam. *In situ* powder X-ray diffraction studies of the Ni– $\text{Bl}_3$  system showed that the crystalline nickel borides,  $\text{Ni}_4\text{B}_3$  and  $\text{Ni}_2\text{B}$ , start to form at temperatures as low as 700 K and 877 K, respectively, which is significantly lower than the typical synthesis temperatures required to produce these borides.  $\text{Ni}_3\text{B}$  synthesized by this method was tested as a supporting material for oxygen evolution reaction (OER) in acidic media. Composite electrocatalysts of  $\text{IrO}_2/\text{Ni}_3\text{B}$  with only 50% of  $\text{IrO}_2$  exhibit current densities and stability similar to pure  $\text{IrO}_2$  at mass loadings lower than  $0.5 \text{ mg cm}^{-2}$ , indicating  $\text{Ni}_3\text{B}$  could be a promising supporting material for acidic OER.

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## Introduction

Metal borides ( $\text{M}_x\text{B}_y$ ) have found use in a wide variety of applications including  $\text{Fe}_2\text{B}$  coatings,<sup>1</sup> {V, Cr, Mn, Fe, Co, Ni, Mo}–B catalysts for electrocatalytic hydrogen evolution reaction (HER) and oxygen evolution reaction (OER),<sup>2–6</sup>  $\text{LaB}_6$  as a hot cathode,<sup>7</sup> and  $\text{VB}_2$  anodes for primary batteries<sup>8</sup> to name a few. Robust and facile solid state synthesis methods for metal borides are, however, hindered by the inertness of boron and, oftentimes, the metal as well. Traditional methods to synthesize

metal borides include arc-melting of the elements, borothermic reduction of metal oxides, or metal flux assisted synthesis.<sup>5</sup> These methods require high temperatures over 1273 K and often require further purification due to the presence of oxygen or other contaminants from the flux or reaction vessel.

The most common lower temperature synthesis of early transition metal borides was discussed by Rao and co-workers in 1995.<sup>9</sup> This study was restricted to the synthesis of group 4–6 metal diborides, in which metal chloride precursors were reacted with  $\text{MgB}_2$  at 1123 K. This method was recently extended by Gillan *et al.* to late transition metal borides.<sup>10</sup> The syntheses of hafnium diboride  $\text{HfB}_2$  and tungsten borides  $\text{WB}_x$  were performed by the reaction of a resistive heated metal wire (Hf or W) with boron bromide instead of using a metal halide precursor.<sup>11,12</sup> This method required a sophisticated apparatus to deal with the liquid boron bromide source and the Hf or W metal wire was only partially converted into boride, thus preserving a metallic core. Fokwa and co-workers developed a straightforward synthesis of nanoscale transition metal borides by reacting anhydrous metal chlorides and elemental boron in Sn flux, which served simultaneously as a high-temperature solvent and as a reducing agent.<sup>13</sup> Another interesting approach to forming Li-containing borides was recently developed using LiH as a precursor.<sup>14</sup> Despite the successful formation of metal borides, these methods are generally limited in that they only produce powders of metal borides (even when

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using a flux or metal hydrides) or they leave behind unreacted elemental metal from the metal source.

While lighter boron halides have already been employed to synthesize metal borides, boron triiodide ( $\text{BI}_3$ ) has the additional benefit of being a solid, greatly increasing the ease of handling it.  $\text{BI}_3$  thermally decomposes at a moderate temperature under reduced pressure and was shown to transport pure boron *via* heat gradient.<sup>17</sup> In addition, unreacted  $\text{BI}_3$  hydrolyzes into  $\text{H}_3\text{BO}_3$  and  $\text{HI}$ , which can be easily dissolved in water.<sup>18</sup> In the present work, motivated by the lack of more efficient methods for making metal borides, we developed a novel borodization route by reaction of metals with  $\text{BI}_3$ . By controlling the amount of  $\text{BI}_3$ , several single-phase transition metal borides, namely  $\text{Fe}_2\text{B}$ ,  $\text{Co}_2\text{B}$ ,  $\text{Ni}_3\text{B}$ ,  $\text{TiB}_2$ ,  $\text{VB}_2$ , and  $\text{CrB}_2$  were synthesized at relatively low temperature of 1073 K. *In situ* powder X-ray diffraction experiments were conducted to understand the formation of the metal borides starting from Ni and  $\text{BI}_3$  reagents. It was found that the crystalline metal borides starts to form at a relatively low temperatures ( $\sim 700$  K), supporting the efficiency of this synthesis. Another advantage of this method is the ability to preserve the morphology of reactant metal after borodization, as we have demonstrated for Ni foam. Topotactic borodization reactions might be useful for converting not only powders, but films and nanocrystals of metals into borides, thus opening avenues to study applications of metal boride phases of various shapes and structures. In addition, topotactic conversion of metal foams into their respective metal phosphide foam electrocatalysts was shown to be of high importance for electrochemical HER and OER.<sup>15–17</sup>

One of the possible applications of metal borides is as a supporting material for electrocatalytic reactions. Acidic OER imposes substantial challenges regarding materials stability and most non-oxide 3d transition metals compounds are prone to severe degradation.<sup>20</sup> However, metal borides show good OER stability in general and treating  $\text{Ni}_3\text{B}$  powders with concentrated hydrochloric acid results in no detectable changes, making such borides good candidates for use in acidic conditions.<sup>21,22</sup> In this work, we investigated the use of  $\text{Ni}_3\text{B}$  as a support for oxygen evolution reaction (OER) electrocatalysis in acidic media. Composite electrocatalysts with different  $\text{IrO}_2/\text{Ni}_3\text{B}$  ratios were tested towards OER; this revealed that the  $\text{Ni}_3\text{B}$  has low capacitance and can be used to effectively support  $\text{IrO}_2$  electrocatalysts. Importantly,  $\text{IrO}_2/\text{Ni}_3\text{B}$  composite electrocatalysts show remarkable specific activity, comparable to that of pure  $\text{IrO}_2$ . Moreover, stability tests showed that the  $\text{IrO}_2/\text{Ni}_3\text{B}$  composite electrocatalyst has a similar stability to pure  $\text{IrO}_2$  during OER electrocatalysis in acidic media. These results suggest that the addition of  $\text{Ni}_3\text{B}$  as a supporting material can be beneficial to reduce use of unsustainable  $\text{IrO}_2$  while retaining most of the original OER activity.

## Experimental part

**Caution:**  $\text{BI}_3$  sublimates at low temperatures and can develop substantial autogenic vapor pressure, which may lead to rupturing and shattering of the reaction ampoule. Amounts of  $\text{BI}_3$  should be kept at a minimum. For the given volume of the sealed ampoule

( $16\text{--}19\text{ cm}^3$ ) at a reaction temperature of 1073 K, the amount of  $\text{BI}_3$  per reaction was limited to less than 650 mg to prevent any potential explosions. Additionally, it is recommended to wrap the ampoule in silica wool or to place it into a secondary container (alumina beaker) filled with sand. In either case, the furnace should be placed in a well-ventilated space, such as a fumehood.

### Synthesis of metal boride bulk powders

Boron triiodide  $\text{BI}_3$  (Sigma Aldrich, 99.8%) and metal powders, namely Ti (Alfa Aesar, 99.9%), V (Alfa Aesar, 99.5%), Cr (Alfa Aesar, 99.5%), Fe (Sigma Aldrich, 99.99%), Co (Alfa Aesar, 99.8%), Ni (Alfa Aesar, 99.996%), La (Alfa Aesar, 99.9%), and Ru (Alfa Aesar, 99.95%) were used as received and handled inside an argon glovebox. Reagents were sealed into 17 mm ID (19 mm OD) silica ampoules with a 14 mm  $\times$  16 mm silica plug for an enclosed length of roughly 9 cm. The amount of  $\text{BI}_3$  per reaction was limited to less than 650 mg. For  $\text{VB}_2$ ,  $\text{CrB}_2$ , and  $\text{Ni}_3\text{B}$  foam reactions, 9 mm ID (11 mm OD) silica tubes were used and the amount of  $\text{BI}_3$  was downscaled to less than  $40\text{ mg cm}^{-3}$  of the sealed ampoule internal volume. The ampoules with reagents were cooled in an ice bath as they were evacuated to  $\approx 10^{-2}$  mbar and then flame sealed under static vacuum. Starting ratios of reagents for selected borides are listed in Table 1.

The sealed ampoule was then laid on its side in a muffle furnace to achieve a slight temperature gradient with the reagents pointing to the back of the furnace where it would be slightly hotter. The default heating profile for the reactions was a 2 h ramp to 1073 K and annealing at that temperature for 24 h. After annealing, the sample was opened in the air (unless specified elsewhere) and washed in distilled water for 2 days. The product was then filtered using a nylon membrane filter and air dried. The Fe reaction was further washed in HCl solution (36–38%, Fisher Scientific) for a day to remove the  $\text{FeO}(\text{OH})$  by-product and finally filtered using a polypropylene membrane filter.

### Topotactic Ni foam transformation

A cut piece of Ni foam (110 pores per inch, 0.3 mm thick, Heze Jiaotong), approximately  $7 \times 17\text{ mm}^2$ , was used instead of Ni powder to synthesize  $\text{Ni}_3\text{B}$ .  $\text{BI}_3$  was loaded as powder with the ratio shown in Table 2 and the Ni foam was inserted vertically into the 9 mm ampoule. The above-described ratio and temperature profile was used.

Table 1 Loaded molar ratios for select metal borides

Targeted boride	Metal : $\text{BI}_3$ ratio
$\text{TiB}_2$	1 Ti : 1 $\text{BI}_3$
$\text{VB}_2$	3 V : 2 $\text{BI}_3$
$\text{CrB}_2$	1 Cr : 1 $\text{BI}_3$
$\text{Fe}_2\text{B}$	7 Fe : 2 $\text{BI}_3$
$\text{Co}_2\text{B}$	7 Co : 2 $\text{BI}_3$
$\text{Ni}_3\text{B}$	7 Ni : 2 $\text{BI}_3$



**Table 2** Molar ratio of Ni : BI<sub>3</sub> starting reagents and corresponding products determined by powder XRD (see Fig. S1). Bold row indicates optimized ratio of starting materials

Ni : BI <sub>3</sub> molar ratio	Crystalline reaction products detected by XRD
1 : 3	Ni <sub>6</sub> BSi <sub>2</sub>
1 : 1	Ni <sub>6</sub> BSi <sub>2</sub> + Ni <sub>3</sub> B + Ni <sub>2</sub> B
4 : 2	Ni <sub>3</sub> B + Ni <sub>2</sub> B
5 : 2	Ni <sub>3</sub> B + Ni <sub>2</sub> B
<b>7 : 2</b>	<b>Ni<sub>3</sub>B</b>
9 : 2	Ni + Ni <sub>3</sub> B
11 : 2	Ni + Ni <sub>3</sub> B

### Powder X-ray diffraction (XRD)

Powder XRD experiments were performed using a 600 Miniflex diffractometer (Rigaku) equipped with Cu-K $\alpha$  radiation and a Ni-K $\beta$  filter. Samples were ground in an agate mortar and pestle, and then deposited on a thin layer of grease on a zero-background Si holder to minimize preferred orientation of particles.

### Synchrotron *in situ* powder XRD

Ni powder and BI<sub>3</sub> crystals totaling 50 mg were ground together with an agate mortar and pestle inside the glovebox. The mixture was loaded into a silica capillary (0.5 mm ID/0.7 mm OD, Friedrich & Dimmock, Inc.). The capillary was then evacuated, and flame sealed. Beamline 17-BM-B at the Advanced Photon Source at Argonne National Laboratory was used for *in situ* experiments. The sample was loaded into a flow cell, equipped with electric heaters, and exposed to synchrotron radiation ( $\lambda = 0.24125$  Å). Data were collected as the sample was heated at 30 K min<sup>-1</sup> to 573 K, ramped at 10 K min<sup>-1</sup> to 1073 K, cooled at 20 K min<sup>-1</sup> to 573 K, and then cooled at 50 K min<sup>-1</sup> to 348 K. Analysis of data was performed using GSAS-II software package.<sup>23</sup>

### Scanning electron microscopy (SEM) and energy dispersive X-ray spectroscopy (EDX)

Elemental analyses were conducted using either a Quanta 250 field emission-SEM (FEI), equipped with a X-Max 80 EDX detector (Oxford), or a JSM-IT200 (JEOL) with built-in EDX detector. Powder samples were loaded into a graphite die, which was then filled with epoxy and polished smooth. A thin conductive carbon film was then deposited on the surface of the puck.

Ni foam and Ni<sub>3</sub>B foam were loaded on carbon tape without further treatments. To process the data, the Aztec and SMILE VIEW software were used for FEI and JEOL, respectively.

### Scanning transmission electron microscopy (STEM)

HAADF-STEM images and Electron Diffraction (ED) data were acquired using JEM ARM200F cold FEG double aberration corrected microscope operated at 200 kV, equipped with a large angle CENTURIO EDX detector, Orius CCD camera, and Quantum GIF. STEM samples were prepared by depositing

ethanolic suspension of the target phase on the Cu holey carbon grid.

### Preparation of IrO<sub>2</sub>/Ni<sub>3</sub>B composite electrocatalysts

Synthesized phase-pure Ni<sub>3</sub>B and commercial IrO<sub>2</sub> (Alfa Aesar, 99.99%) powders were mixed in 1 : 2, 1 : 1, and 2 : 1 molar ratios with a total amount of each mixture being 150 mg; the corresponding samples were labeled as 25%, 50%, and 75% of Ni<sub>3</sub>B. Once the samples were combined, they were ball-milled using a SPEX mixer/mill 8000M for an hour to ensure that the powders were completely blended and became fine enough for facile electrocatalytic ink formulation. For comparison purposes, samples of pure Ni<sub>3</sub>B and pure IrO<sub>2</sub> were also ball-milled in a similar manner. The milled powder was then drop-cast on a home-made glassy carbon (GC) rotating disk electrode (RDE) with a geometric area of 0.2826 cm<sup>2</sup>, which served as the working electrode (WE) for electrochemical characterizations.

The ink containing the electrocatalysts was prepared by mixing water, isopropanol (volume ratio of water : isopropanol = 3 : 1), Nafion ionomer solution (5% in aliphatic alcohols and water, Sigma-Aldrich), and the composite electrocatalyst powders (50 mg mL<sup>-1</sup>). Before drop-casting, the GC RDE was polished with alumina powder for 5 min and then rinsed with water and ethanol. This procedure was repeated 3 times. After an ultrasonic treatment for 15 min, the as-formulated ink was drop-casted on the GC RDE and then left in a solvent-vapor-saturated (water + isopropanol) atmosphere overnight for drying. The mass loading of IrO<sub>2</sub>/Ni<sub>3</sub>B (0.25, 0.5, 1.0, 2.0 mg cm<sup>-2</sup>) was controlled by the volume of the drop-casted ink.

### Characterization of OER activity and stability in acidic media

The electrochemical characterization of OER activity and stability was done in a conventional three-electrode glass cell. The counter electrode (CE) and reference electrode (RE) was a GC rod and a saturated Ag|AgCl electrode, respectively. All potentials are quoted in reference to the reversible hydrogen electrode (RHE). The electrolyte solution was 0.5 M H<sub>2</sub>SO<sub>4</sub> prepared from H<sub>2</sub>SO<sub>4</sub> (Merck, 99.999%) and ultrapure water (18.2 MΩ cm, Milli-Q). All the experiments were performed under N<sub>2</sub> (5.0, LindeGas) saturated conditions. The applied potential was controlled by PGSTAT302N potentiostat (Metrohm) and the RDE rotating rate by an Autolab RRDE rotator (Metrohm). Before the electrochemical characterization, the WE (with electrocatalysts) was carefully rinsed with pure water, followed by electrochemical cleaning for 15 min by performing cyclic voltammetry from 0.05 to 1.1 V<sub>RHE</sub> at a scan rate of 200 mV s<sup>-1</sup>. All the data shown was corrected for a voltage drop (*iR*) of 80% unless specified elsewhere.

## Results and discussion

### Synthesis

We have shown that the refractory boride, BP, can be synthesized using BI<sub>3</sub> as the source of boron.<sup>24</sup> Later, refractory metal borides such as HfB<sub>2</sub> and WB<sub>2</sub> have been prepared by reaction



of the resistive heated metal wire with  $\text{BBr}_3$  vapor.<sup>11,12</sup> In the current work, we extend  $\text{BI}_3$ -assisted synthesis to first row transition metals. Similar to solid state metathesis reactions, the main driving force for this reaction is the formation of thermodynamically stable, water-soluble metal iodides:  $(nx + 3)\text{M} + n\text{BI}_3 \rightarrow n\text{M}_x\text{B} + 3\text{MI}_n$  ( $x = 1/6$  to  $3$ ;  $n = 2, 3$ , and  $4$ ). For the selected metals ( $\text{M} = \text{Ti}, \text{V}, \text{Cr}, \text{Fe}, \text{Co}$ , and  $\text{Ni}$ ) phase-pure polycrystalline metal boride powders or samples with minute amounts of admixtures (Fig. 1) can be successfully synthesized using the optimized ratios of the precursors, as described in Table 1.  $\text{Co}_2\text{B}$ ,  $\text{Fe}_2\text{B}$ , and  $\text{VB}_2$  can be synthesized by a balanced reaction, *i.e.*, no excess of metal or halide is expected in the products:  $7\text{Co} + 2\text{BI}_3 \rightarrow 2\text{Co}_2\text{B} + 3\text{CoI}_2$ . In turn,  $\text{Ni}_3\text{B}$ ,  $\text{CrB}_2$ , and  $\text{TiB}_2$  required an excess of  $\text{BI}_3$ , indicating a formation of the gaseous iodine by-product:  $7\text{Ni} + 2\text{BI}_3 \rightarrow 2\text{Ni}_3\text{B} + \text{NiI}_2 + 2\text{I}_2 \uparrow$ ,  $4\text{Cr} + 4\text{BI}_3 \rightarrow 2\text{CrB}_2 + 2\text{CrI}_3 + 3\text{I}_2 \uparrow$ , and  $2\text{Ti} + 2\text{BI}_3 \rightarrow \text{TiB}_2 + \text{TiI}_4 + \text{I}_2 \uparrow$ . Note that the melting points of the corresponding borides, 1429 K ( $\text{Ni}_3\text{B}$ ) and 3498 K ( $\text{TiB}_2$ ) are significantly higher than the utilized reaction temperature, 1073 K.

For Ni, the reaction of 7 equivalents of Ni with 2 equivalents of  $\text{BI}_3$  provided single phase samples of  $\text{Ni}_3\text{B}$ . These reaction ratios were utilized for the topotactic conversion of Ni foam, as well as for the preparation of the electrocatalysts discussed below. Additionally, a detailed study of the impact of metal to  $\text{BI}_3$  ratio was performed. Employing a large excess of  $\text{BI}_3$  resulted in the formation of  $\text{Ni}_6\text{BSi}_2$  due to reaction with the silica ampoule. An unknown amorphous phase was detected in the reactions that produced  $\text{Ni}_6\text{BSi}_2$ , possibly also resulting from reaction with the silica ampoule. A stoichiometric reaction of

$9\text{Ni} : 2\text{BI}_3$  or reactions with higher Ni contents resulted in the presence of leftover metallic Ni in the products. As detailed in Table 2,  $\text{Ni}_2\text{B}$  could be synthesized as well, in the presence of a slight excess of  $\text{BI}_3$ , with  $\text{Ni}_3\text{B}$  as a side product (Fig. S1†).

Other metals, namely La and Ru, were investigated using starting ratios for balanced reactions (*i.e.*,  $\text{La} : \text{BI}_3 = 7 : 6$  and  $\text{Ru} : \text{BI}_3 = 3 : 2$ ), assuming formation of  $\text{LaI}_3$  and  $\text{RuI}_2$  co-products.  $\text{Ru}_7\text{B}_3$  and  $\text{LaB}_6$  were successfully synthesized using the heating profile described in the experimental section, although an admixture of unreacted metal was present (Fig. S2†).  $\text{CrB}$  was also successfully synthesized using the same stoichiometric ratio with a single unidentified peak in the PXRD pattern at  $2.44 \text{ \AA}^{-1} Q$  (Fig. S3†). Further optimizations of reagents ratios and temperature profiles are required to produce phase-pure samples of these borides.

The herein developed method for synthesis of metal borides demonstrates universality by allowing for formation of materials with high diversity in B : M ratios, crystal structures, and degrees of B–B bonding (Scheme 1). As the B : M ratio decreases from 6 to  $1/3$ , the dimensionality of the boron framework changes from 3D to 0D. For  $\text{LaB}_6$ , the 3D boron framework is composed of covalently-linked octahedra. 2D graphene-like boron layers are present in the crystal structures of  $\text{MB}_2$  ( $\text{TiB}_2$ ,  $\text{VB}_2$ , and  $\text{CrB}_2$ ), while 1D boron chains are found in MB ( $\text{CrB}$ ) and  $\text{M}_2\text{B}$  ( $\text{Co}_2\text{B}$  and  $\text{Fe}_2\text{B}$ ). Further increasing the metal content stabilizes structures with isolated boron atoms surrounded by  $\text{M}_6$  trigonal prisms (as in  $\text{Ru}_7\text{B}_3$ ) or  $\text{M}_9$  tricapped trigonal prism (as in  $\text{Ni}_3\text{B}$ ).

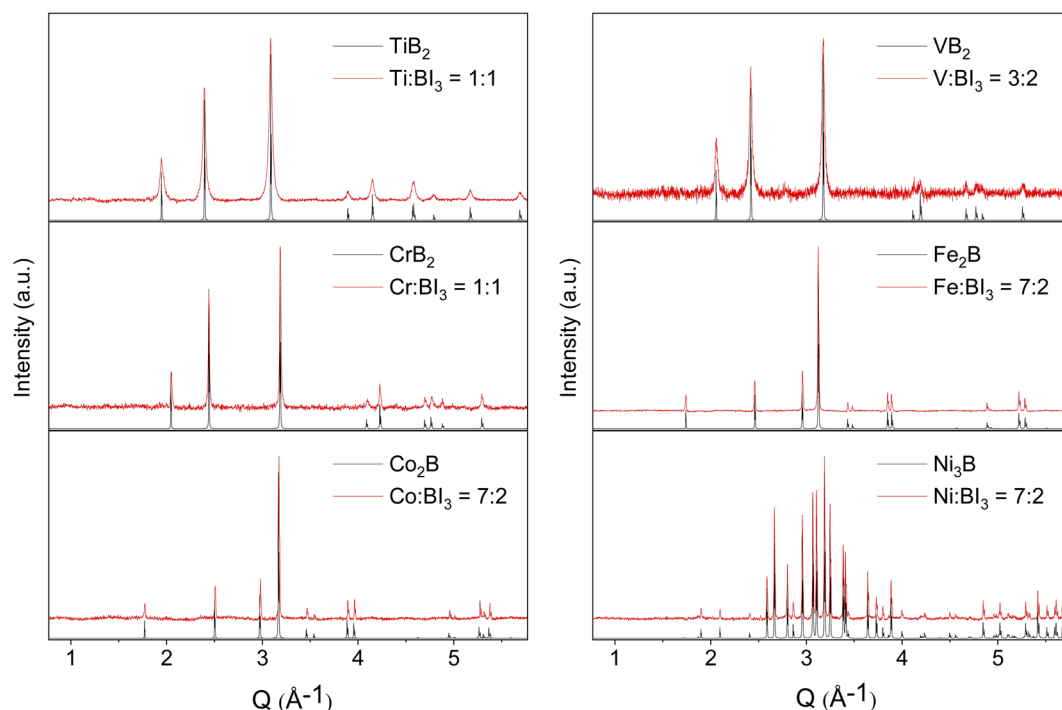
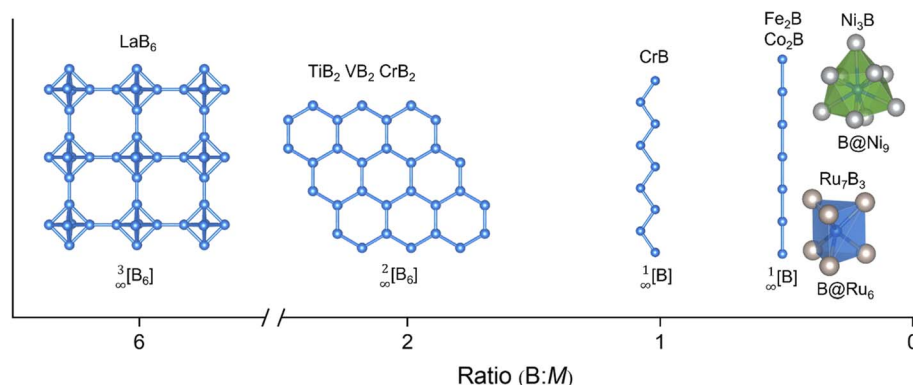


Fig. 1 Calculated (black) and experimental (red) powder XRD patterns of  $\text{TiB}_2$ ,  $\text{VB}_2$ ,  $\text{CrB}_2$ ,  $\text{Fe}_2\text{B}$ ,  $\text{Co}_2\text{B}$ , and  $\text{Ni}_3\text{B}$  synthesized through the reaction of metal powders and  $\text{BI}_3$ . A peak at  $2.79 \text{ \AA}^{-1} Q$  present in all patterns, but most visible in the case of the  $\text{VB}_2$  sample, is attributed to contribution from the grease used for attaching powders to the XRD holder.







Scheme 1 Boron frameworks in the metal borides synthesized in this study with respect to the ratio of boron to metal. Boron: blue; metal: grey.

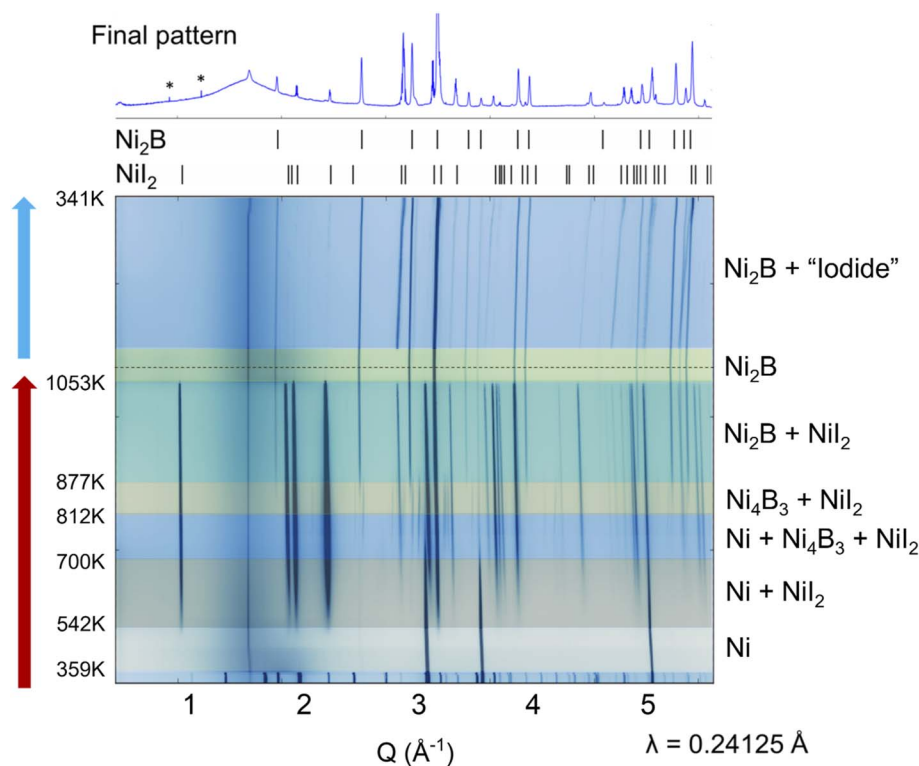


Fig. 2 Contour plot of *in situ* powder XRD patterns upon reaction of Ni and  $\text{BI}_3$ . The red arrow on the left indicates heating while the blue one corresponds to the cooling process. On top is the room temperature powder XRD pattern after completion of the *in situ* study, with tick marks corresponding to calculated positions for  $\text{Ni}_2\text{B}$  and  $\text{NiI}_2$  diffraction peaks. Asterisks mark instrumental artifacts in the powder pattern.

### *In situ* powder XRD

To understand the mechanism of borodization, *in situ* experiments were performed to study the formation of nickel boride (Fig. 2). The provided temperatures are approximate ( $\pm 20$  K) because the thermocouple is located outside the reaction capillary. Nevertheless, the series of phase transformations can still be determined with high reliability which further aids synthesis efforts and our knowledge of solid state reaction mechanisms.<sup>25–28</sup> *In situ* XRD studies were previously shown to be useful for synthesis of novel boron-containing compounds.<sup>29</sup> In the studied synthesis of nickel boride,  $\text{BI}_3$  amorphized or went into the gaseous phase at 359 K, leaving only Ni to be

detected by powder XRD. At 542 K,  $\text{NiI}_2$  is formed. New diffraction peaks appear in the powder patterns from 700 K to 877 K, which can be assigned to  $\text{Ni}_4\text{B}_3$ . It should be mentioned that the  $\text{Ni}_4\text{B}_3$  intermediate was detected in other high-temperature *in situ* XRD studies involving the decomposition of ternary Li–Ni–B compounds.<sup>30</sup> The detection of  $\text{Ni}_4\text{B}_3$  demonstrates that the formation of  $\text{NiI}_2$  provides a strong enough driving force to make crystalline nickel boride at extremely low temperatures of  $\sim 700$  K.  $\text{Ni}_4\text{B}_3$  may serve as an intermediate in the studied reaction. One can assume that an amorphous Ni–B phase is present in the capillary as well, at least in the 810–875 K range because nearly all the elemental Ni



has been consumed by 812 K, and the formation of  $\text{Ni}_2\text{B}$  is detected only at 877 K.  $\text{Ni}_2\text{B}$  is the final thermodynamically stable phase and as soon as it forms, the peaks corresponding to  $\text{Ni}_4\text{B}_3$  can no longer be detected. At 1053 K,  $\text{NiI}_2$  melts, in agreement with its reported melting point, and  $\text{Ni}_2\text{B}$  is the only crystalline phase present. Similarly, upon cooling, only  $\text{Ni}_2\text{B}$  and an unknown phase were detected. Intense peaks around  $2 \text{ \AA}^{-1}$  corresponding to  $\text{NiI}_2$  were no longer observed. Given the water solubility of the unidentified byproduct and the observation of  $\text{NiI}_2$  in the products of larger scale *ex situ* reactions, the unidentified phase is hypothesized to contain nickel and iodine. The persistent peaks in the powder patterns at  $\sim 1 \text{ \AA}^{-1}$  are unaffected by temperature; thus these peaks are attributed to the experimental setup and not intrinsic to the studied system. The highly electrostatic nature of  $\text{BI}_3$  and the small scale of the reaction may have prevented accurate loading ratios of the starting materials. Additionally, increased vapor pressure or the lack of temperature gradient in the *in situ* set-up may have contributed to the reaction outcome as  $\text{Ni}_2\text{B}$  rather than  $\text{Ni}_3\text{B}$  as observed in larger scale *ex situ* reactions. Representative powder XRD patterns at different temperatures in the *in situ* study and calculated patterns of crystalline known phase are plotted in Fig. S5.†

### Topotactic reaction

The developed borodization method is suitable for generating metal borides of unique morphologies. A high surface area is

known to be important for electrocatalysts,<sup>16</sup> and Ni foam borodization was therefore attempted. The  $\text{Ni}_3\text{B}$  reaction was repeated using Ni foam (Fig. 3). While a slight amount of  $\text{Ni}_2\text{B}$  was detected by powder XRD, a predominantly  $\text{Ni}_3\text{B}$  foam was successfully synthesized, albeit relatively brittle (Fig. 3d). Energy dispersive X-ray spectroscopy (EDX) showed no trace of iodine, demonstrating all iodide species were dissolved in water during the rinsing process (Fig. S4†). Fig. 3a shows the optical images of unreacted Ni foam (left) and the resultant  $\text{Ni}_3\text{B}$  foam (right) synthesized using  $\text{BI}_3$ . The original morphology of the Ni foam was preserved, and the color of the foam changed from shiny silver to dark brown. SEM images showed that the initial Ni foam was composed of Ni ligaments arranged in a tangled and porous 3D network (Fig. 3b). After borodization, this structure was retained, showing the topotactic ability of the reaction (Fig. 3c). Occasionally, the foam structure was altered during the course of the reaction such that the pores were eliminated (Fig. S6†). This suggests that location of the Ni foam in the ampoule (with or without direct contact with  $\text{BI}_3$ ) might impact the morphology. Increasing the Ni :  $\text{BI}_3$  ratio to 11 : 2 appeared to reduce the brittleness of the resulting  $\text{Ni}_3\text{B}$  foam, although unreacted Ni was expectedly detected by powder XRD. Nevertheless, the as-produced  $\text{Ni}_3\text{B}/\text{Ni}$  foam was still stiff and did not retain the flexibility of the original Ni foam. The aforementioned formation of  $\text{Ni}_6\text{BSi}_2$  can also be detected by EDX as it crystallizes as well-shaped hexagonal plate-like crystals (Fig. S7†).

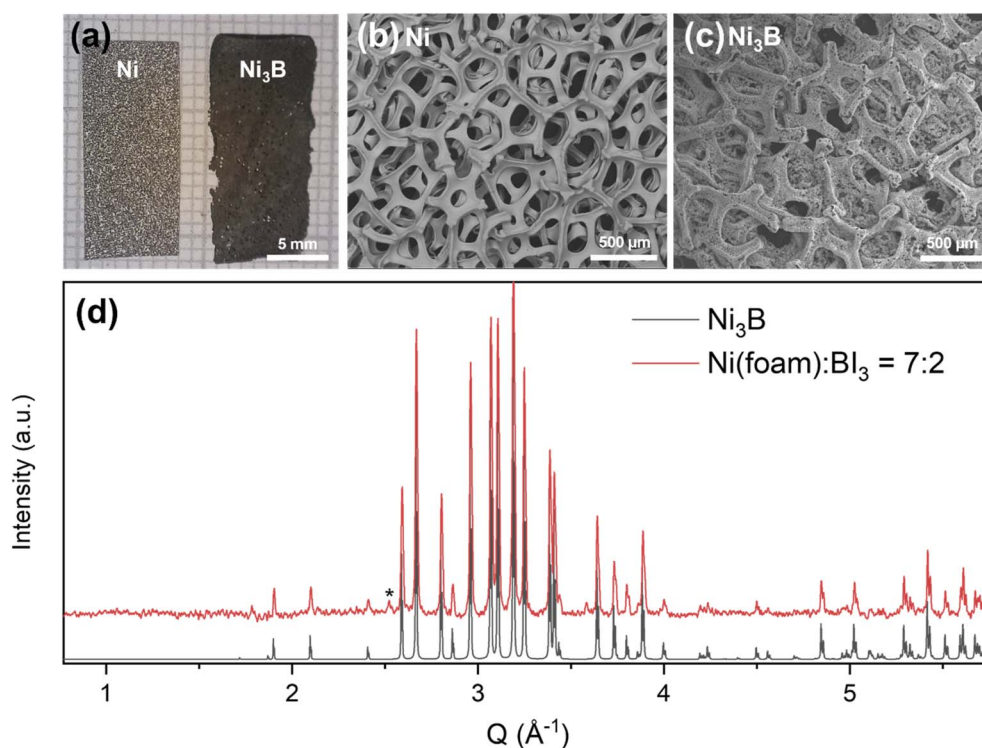


Fig. 3 (a) Photographs of pristine Ni foam (left) used as a reagent and the resultant product  $\text{Ni}_3\text{B}$  foam (right) synthesized by a topotactic borodization. SEM backscattered electron images of (b) pristine Ni foam and (c) the resulting  $\text{Ni}_3\text{B}$  foam. (d) Powder XRD patterns of the ground  $\text{Ni}_3\text{B}$  foam (red) and calculated patterns for  $\text{Ni}_3\text{B}$  (black). A peak at  $2.52 \text{ \AA}^{-1} Q$  marked with \* corresponds to  $\text{Ni}_2\text{B}$  admixture.



### Ternary mixed metal borides

Synthesis of ternary borides were approached *via* a proposed borodization of binary metal alloys. A Ni/Co alloy was produced by grinding a 1 : 1 molar ratio of Ni and Co powders together, which were then loaded in a silica ampoule and heated at 1173 K for 72 h. The resulting alloy ingot could not be crushed and was instead cut into smaller pieces using diagonal cutting pliers. These Ni/Co alloy pieces were then loaded in a 7 : 2 molar ratio of the alloy and  $\text{BI}_3$  in accordance with the optimized  $\text{Ni}_3\text{B}$  synthesis (Table 2). The resulting product was a malleable foil and grey-brown powder. The powder XRD pattern of the latter resembles  $\text{Ni}_3\text{B}$  or  $\text{Co}_3\text{B}$  patterns with  $\sim 0.01 \text{ \AA}^{-1}$   $Q$  peak shifts (Fig. S8†). Analysis by EDX found the ternary boride powder to contain a mixture of Ni and Co with 70/30 molar ratio, *i.e.*  $\text{Ni}_{2.1(1)}\text{Co}_{0.9}\text{B}$  (Fig. S9†). Different areas of the sample were analyzed, and the chemical composition was almost identical for all probed areas. Boron is too light to be accurately determined by EDX, and its signal overlaps with C and O signals, which are ubiquitous in all samples. The metal foil was most likely unreacted NiCo alloy and identified to be  $\sim 1 : 1$  Co : Ni (47.3(3)% Ni and 52.7(6)% Co) which is similar to the nominal composition of the alloy reactant. The synthesis of  $\text{Ni}_{2.1(1)}\text{Co}_{0.9}\text{B}$  suggests that ternary borides may be successfully synthesized by our methodology when using an alloyed precursor with  $\text{BI}_3$ , rather than individual metal elements. In contrast, a controlled reaction starting from a mixture of Fe and Co metal powders and  $\text{BI}_3$  resulted in the formation of two individual borides rather than a ternary boride.

### OER electrocatalysis

Metal borides have been widely explored as electrocatalysts. The *Fokwa* group showed applicability of various metal borides as HER electrocatalysts for  $\text{H}_2$  generation.<sup>5</sup> For OER, different metal borides have been evaluated but these studies are limited to alkaline conditions.<sup>6,19–22,31</sup> Alkaline OER currently has lower technological importance as all current industrial membrane-based electrolyzers operate in acidic media due to restrictions

on ion-conductive membranes.  $\text{IrO}_2$  is state-of-the-art acidic OER catalysts, but is not sustainable due to the scarcity of Ir. Thus, the development of acidic OER electrocatalysts which can reduce the usage of noble and scarce Ir metal is of high relevance.<sup>32</sup>

OER activity was evaluated for composites of  $\text{IrO}_2/\text{Ni}_3\text{B}$  with different ratios by measuring cyclic voltammetry (CV) at a scan rate of  $50 \text{ mV s}^{-1}$  in  $\text{N}_2$  saturated  $0.5 \text{ M H}_2\text{SO}_4$  solutions. As shown in Fig. 4a, the onset potential for OER with pure  $\text{IrO}_2$  with mass loading of  $1 \text{ mg cm}^{-2}$  was  $\approx 1.46 \text{ V}_{\text{RHE}}$ , which is consistent with previous reports.<sup>33–36</sup> When compared with  $\text{IrO}_2$ , the OER activities of pure  $\text{Ni}_3\text{B}$  (mass loading of  $2 \text{ mg cm}^{-2}$ ) were negligible like the bare GC disk, and didn't show any additional reaction before OER (Fig. 4a). For these reasons, we assume that  $\text{Ni}_3\text{B}$  works as a conductive supporting material with negligible activity and that  $\text{IrO}_2$  is the main contributor to the OER activities observed for the composite electrocatalysts.

To allow for quantitative comparison of the acidic OER activity under different conditions, *e.g.*, mass loading or molar ratio of  $\text{Ni}_3\text{B}$ , the OER currents at a constant applied potential were normalized by electrochemical surface area (ECSA). The ECSA of  $\text{IrO}_2$  and  $\text{IrO}_2/\text{Ni}_3\text{B}$  were determined by integral charge ( $Q$ ) from  $0.05 \text{ V}_{\text{RHE}}$  to  $1.0 \text{ V}_{\text{RHE}}$ :

$$\text{ECSA} = \frac{Q_{\text{exp}}}{Q_{\text{ref}}}$$

where the subscripts exp and ref refer to experimental values and reference values (specific charge density,  $Q_{\text{ref}} = 320 \text{ \mu C cm}^{-2}$  for  $\text{IrO}_2$ ), respectively.<sup>37</sup> The ECSA-normalized current density ( $j_{\text{ECSA}}$ ) was calculated by excluding the contribution of GC to both OER current and ECSA:

$$j_{\text{ECSA}} = \frac{i}{\text{ECSA}} = \frac{i_{\text{total}} - i_{\text{GC}}}{(Q_{\text{total}} - Q_{\text{GC}})/Q_{\text{ref}}}$$

where the subscripts total and GC refer to total measured current (or integral charge) for the electrocatalyst drop-casted WE and for the bare GC, respectively. The as-calculated current densities are shown in Fig. 4b. For pure  $\text{IrO}_2$  with

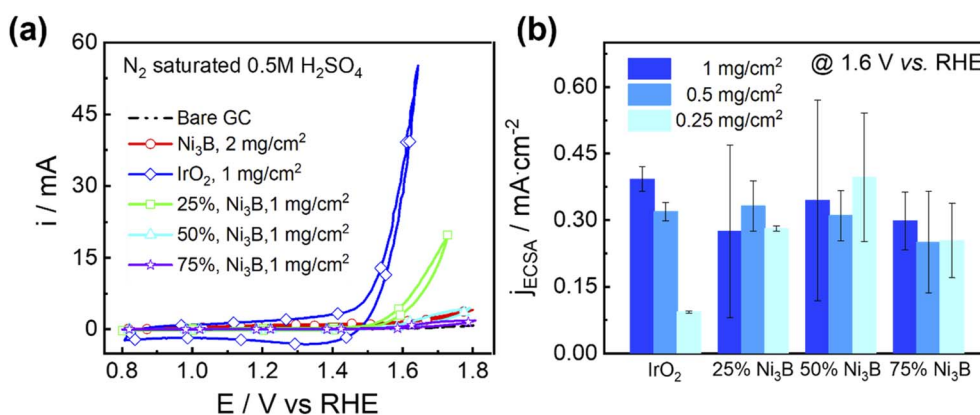


Fig. 4 (a) Cyclic voltammogram of bare GC,  $\text{Ni}_3\text{B}$  (loading =  $1 \text{ mg cm}^{-2}$ ),  $\text{IrO}_2$  (loading =  $1 \text{ mg cm}^{-2}$ ),  $\text{IrO}_2 + \text{Ni}_3\text{B}$  mixture (molar ratio of  $\text{Ni}_3\text{B}$  = 25%, 50%, 75%, loading =  $1 \text{ mg cm}^{-2}$ ) in  $\text{N}_2$  saturated  $0.5 \text{ M H}_2\text{SO}_4$ . Scan rate =  $50 \text{ mV s}^{-1}$ , electrode rotating rate = 1600 rpm. (b) OER current densities of  $\text{Ni}_3\text{B} + \text{IrO}_2$  mixture with different molar percentage of  $\text{Ni}_3\text{B}$  (0/25%/50%/75%) in  $\text{N}_2$  saturated  $0.5 \text{ M H}_2\text{SO}_4$ . The measured currents by CV at  $1.6 \text{ V}$  vs. RHE were normalized by ECSA. Electrode rotation rate = 1600 rpm.



$1 \text{ mg cm}^{-2}$  mass loading, the current density at  $1.6 \text{ V}_{\text{RHE}}$  is  $\approx 0.4 \text{ mA cm}^{-2}$ , which is close to the reported values for polycrystalline  $\text{IrO}_2$ ,<sup>33</sup> single crystalline  $\text{IrO}_2$ ,<sup>34–36</sup> and nanosized  $\text{IrO}_2$  in proton exchange membrane electrolyzers.<sup>37</sup> The uncertainties in  $\text{IrO}_2$  current densities were rather small, which indicates good control of the electrochemical experiments. With the decrease of mass loading per geometrical area from  $1 \text{ mg cm}^{-2}$  to  $0.5 \text{ mg cm}^{-2}$ , the  $j_{\text{ECSA}}$  for  $\text{IrO}_2$  slightly decreases. However, the  $j_{\text{ECSA}}$  for  $\text{IrO}_2$  with  $0.25 \text{ mg cm}^{-2}$  mass loading is  $\sim 25\%$  of that for sample with  $1 \text{ mg cm}^{-2}$  loading.

The normalization by ECSA can omit the influence of the electrolyte accessibility to the electrocatalysts. However, normalization by mass loading, where the electrode accessibility would be taken into account, also presents some challenges,<sup>38</sup> since mass loading can influence the  $Q_{\text{ref}}$  value ( $Q_{\text{ref}}$  for  $\text{IrO}_2$  changes from  $320 \mu\text{C cm}^{-2}$  at mass loadings  $\leq 1 \text{ mg cm}^{-2}$  to  $600 \mu\text{C cm}^{-2}$  at reasonably lower  $\text{IrO}_2$  loading, around  $0.15 \text{ mg cm}^{-2}$  (ref. 37 and 39)). Moreover, lower mass loadings can lead to higher dissolution or deactivation of  $\text{IrO}_2$  during the electrochemical cleaning process, even at potentials more negative than the onset potential of OER.<sup>33</sup> Lastly, the effect of mass loading on particle distribution may originate from different derivations of the electron/ion transport. For these reasons, in the present manuscript, the currents were normalized by ECSA unless stated otherwise.

Fig. 4b shows that for the  $\text{IrO}_2/\text{Ni}_3\text{B}$  composite electrocatalysts, the OER current densities are close to pure  $\text{IrO}_2$ , regardless of the  $\text{Ni}_3\text{B}$  molar ratio and the total loading. This indicates that the addition of  $\text{Ni}_3\text{B}$  helps to reduce the utilization of  $\text{IrO}_2$  while maintaining most of the OER activity. Thus, the synthesized  $\text{Ni}_3\text{B}$  has the potential to become an alternative support to carbon materials, which are unstable at elevated temperatures and require high potentials in electrolyzers.<sup>40–43</sup> The overpotential and Tafel slope of pure  $\text{IrO}_2$  and  $\text{Ni}_3\text{B}/\text{IrO}_2$  composites are shown in Table S1 and Fig. S10.†  $\text{IrO}_2/\text{Ni}_3\text{B}$  samples showed higher uncertainties when compared to the pure  $\text{IrO}_2$  electrocatalyst. To investigate the origin of this uncertainty, contact angle studies for Ni and  $\text{Ni}_3\text{B}$  foam with water were conducted (Fig. S11†). Results showed that  $\text{Ni}_3\text{B}$  foam became hydrophilic compared to the hydrophobic precursor, Ni foam, indicating that the uncertainty of the composite may come from the deactivation of  $\text{Ni}_3\text{B}$ , not from low interaction between the catalyst and electrolyte.

The stability of the composite electrocatalysts in acidic OER was evaluated using CV under RDE configuration from  $0.8 \text{ V}_{\text{RHE}}$  to  $1.8 \text{ V}_{\text{RHE}}$  at a scan rate of  $50 \text{ mV s}^{-1}$ . The obtained results are plotted in Fig. 5. For the comparison, we plotted the OER current, ECSA, and OER current densities at different cycle numbers normalized to the first cycle (*i.e.*, first cycle shows 100%). In this way, we can access the relative deactivation and/or changes caused by the increase in time (number of measurements). The OER current densities were normalized by the ECSA determined by the data obtained from the same cycle. In Fig. 5, the OER currents, ECSA, and OER current densities of all the tested electrocatalysts show a significant reduction with increasing the number of scanning CV cycles. We have carefully confirmed that no obvious electrocatalyst detachment from the

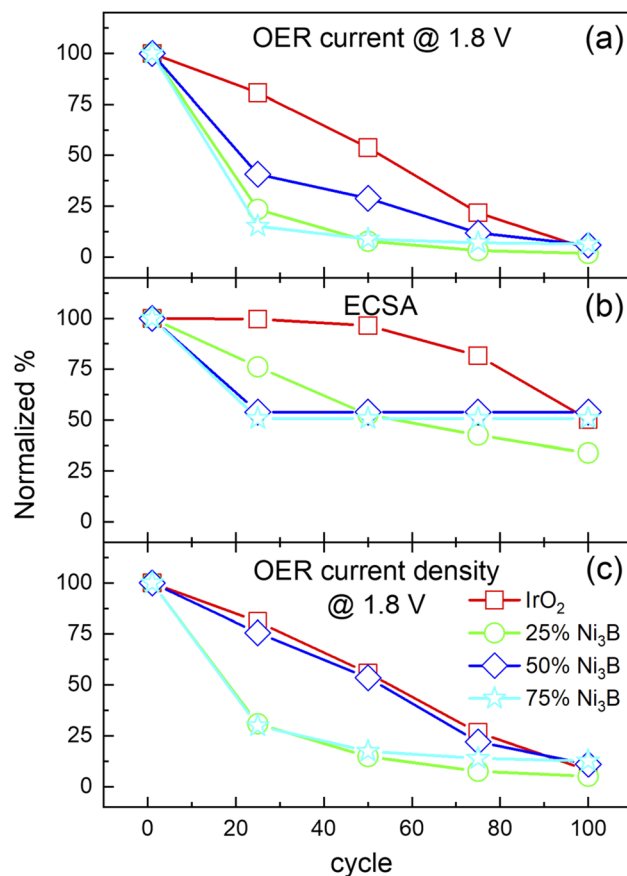


Fig. 5 Normalized (a) OER current at  $1.8 \text{ V}_{\text{RHE}}$ , (b) ECSA and (c) current density at  $1.8 \text{ V}_{\text{RHE}}$  for  $\text{IrO}_2/\text{Ni}_3\text{B}$  composite electrocatalyst under different scanning cycles from  $0.8 \text{ V}_{\text{RHE}}$  to  $1.8 \text{ V}_{\text{RHE}}$  (0/25%/50%/75%  $\text{Ni}_3\text{B}$ ). The data is normalized such that the value at the first cycle was set to 100%. Scan rate =  $50 \text{ mV s}^{-1}$ . Electrolyte solution:  $\text{N}_2$  saturated  $0.5 \text{ M H}_2\text{SO}_4$ , electrode rotating rate =  $1600 \text{ rpm}$ . Mass loading =  $1 \text{ mg cm}^{-2}$ .

GC or dropping into the electrolyte occurred during the whole measurement.

The observed OER deactivation on  $\text{IrO}_2$  has been previously attributed to an artifact caused by the insufficient removal of  $\text{O}_2$  bubbles in the porous electrode layer.<sup>44</sup> The displacement of electrolyte within or near the electrocatalyst layer by  $\text{O}_2$  will lead to possible loss of ionic contact and a substantial reduction of the ECSA, as confirmed by Fig. 5b. In this regard, hereon, we will use the expression “apparent” stability. For  $\text{IrO}_2/\text{Ni}_3\text{B}$  composite electrocatalysts with different molar ratios, the observed behavior was different. In the case of 50%  $\text{Ni}_3\text{B}$ , the “apparent” stability was found to be similar to that of pure  $\text{IrO}_2$ , while in the cases of 25% and 75%  $\text{Ni}_3\text{B}$ , the “apparent” stability decreased. Since the  $\text{IrO}_2/\text{Ni}_3\text{B}$  samples showed no difference in hydrophobicity or conductivity (results not shown), this behavior may originate from a combination of two hypothesized factors that operate in opposite directions:

(1)  $\text{Ni}_3\text{B}$  shows some deactivation even in the absence of  $\text{IrO}_2$  (Fig. S12†). Therefore, the addition of  $\text{Ni}_3\text{B}$  into  $\text{IrO}_2$  will introduce another component to the stability of the layer.





Considering this factor alone, the higher the  $\text{Ni}_3\text{B}$  molar ratio, the worse the “apparent” stability.

(2) Since  $\text{IrO}_2$  is more active toward acidic OER than pristine  $\text{Ni}_3\text{B}$  (Fig. 4a), the addition of  $\text{Ni}_3\text{B}$  to  $\text{IrO}_2$  will reduce the total OER current, and reduce  $\text{O}_2$  generation and accumulation within the electrocatalyst's layers. Considering this factor alone, the higher the  $\text{Ni}_3\text{B}$  molar ratio, the better the “apparent” stability.

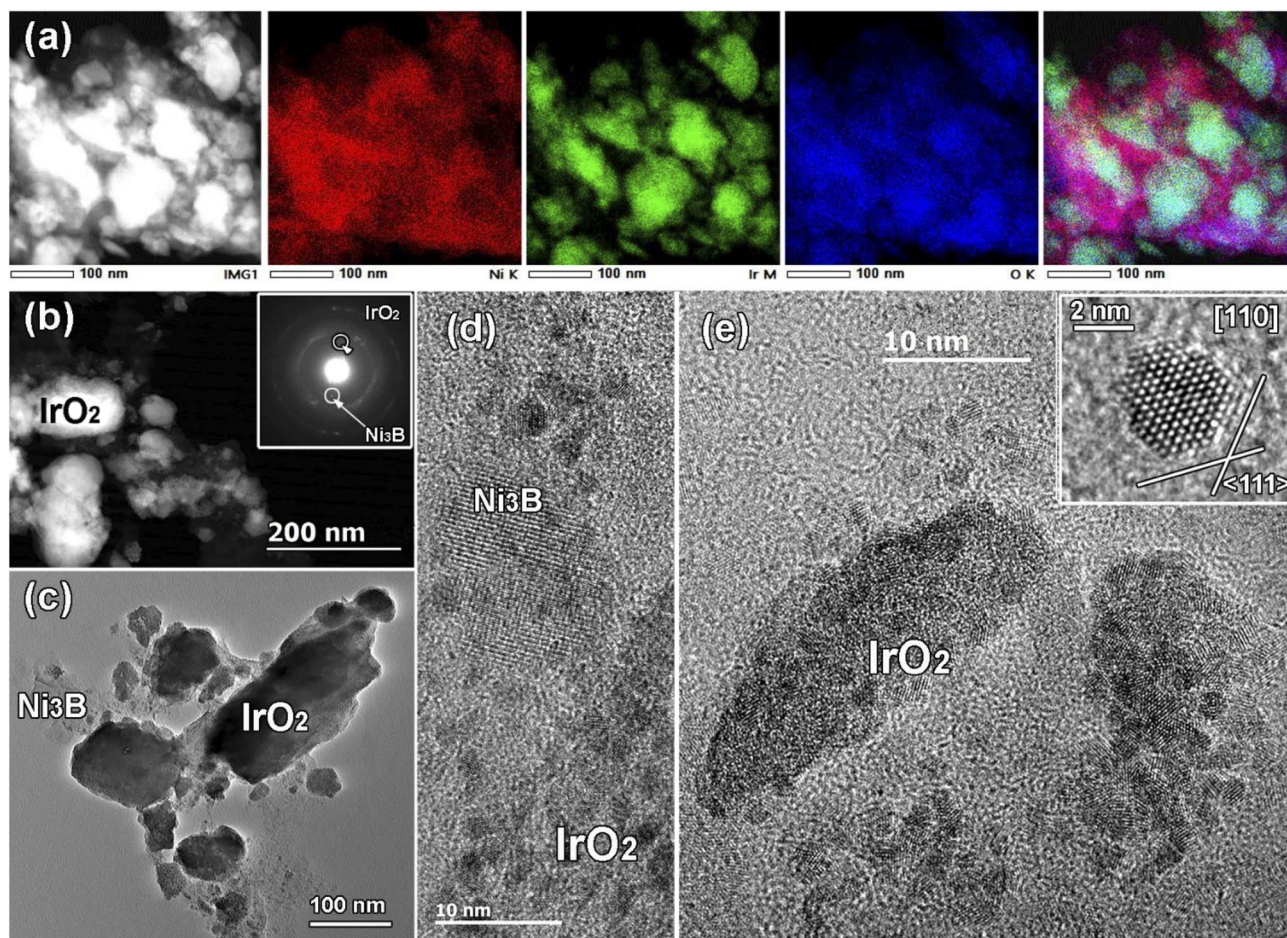
Therefore, the best “apparent” stability was achieved at 50%  $\text{Ni}_3\text{B}$  molar ratio, which can be explained by a good balance between (1) the amount of support and (2) the reducing  $\text{O}_2$  production and consequent accumulation in the electrocatalyst layer. Further measurements are necessary to quantitatively assess these two competitive factors, which is out of the scope of this work.

To investigate potential degradation of the catalysts during the OER tests, a detailed scanning-transmission electron microscopy study was performed. A freshly prepared  $\text{IrO}_2/\text{Ni}_3\text{B}$  composite samples (Fig. 6) showed a good dispersion of Ir oxide

and Ni boride crystallites without extreme agglomeration. Both  $\text{IrO}_2$  and  $\text{Ni}_3\text{B}$  demonstrated high crystallinity.

A sample taken from the same batch was used for aforementioned catalysis tests and afterwards subjected to a microscopy investigation (Fig. 7). EDX-STEM mapping showed a similar dispersion of Ir oxide and Ni boride before and after catalysis (compare Fig. 6a, b and 7a, b) indicating no significant sample degradation. An important difference is a higher degree of agglomeration of  $\text{IrO}_2$  nanocrystals was detected after OER. The observed aggregation might be another potential mechanism for a catalyst's deactivation. To probe this, methods other than ball-milling to prepare composites should be investigated, such as chemical precipitation of  $\text{IrO}_2$  nanocrystals on  $\text{Ni}_3\text{B}$ , which is a subject for further study. The surface of nickel boride may be oxidized during OER reaction, nevertheless the crystalline  $\text{Ni}_3\text{B}$  was still present in the sample after catalysis as shown by SAED (Fig. 7d inset).

Overall, we have shown the potential of  $\text{Ni}_3\text{B}$  as a potential supporting material for acidic OER. The addition of 50%  $\text{Ni}_3\text{B}$



**Fig. 6** TEM results of  $\text{Ni}_3\text{B}/\text{IrO}_2$  sample before the catalysis test. (a) Low magnification of high-angle annular dark-field (HAADF)-STEM image and simultaneously acquired EDX-STEM elemental mapping for N-K, Ir-M, O-K edges and overlaid color images. (b) Low magnification HAADF-STEM image and selected area electron diffraction (SAED) pattern is given as inset. (c) Bright field low magnification TEM image showing the agglomeration of  $\text{IrO}_2$  nanoparticles (dark contrast) and  $\text{Ni}_3\text{B}$  support (light contrast); (d) HRTEM image of  $\text{IrO}_2$  nanoparticles on  $\text{Ni}_3\text{B}$  support material; (e) Bright field HRTEM image of agglomeration of  $\text{IrO}_2$  nanoparticles. Inset: [110] HRTEM image of separated  $\text{IrO}_2$  nanoparticle faceted along the  $\langle 111 \rangle$  plane.





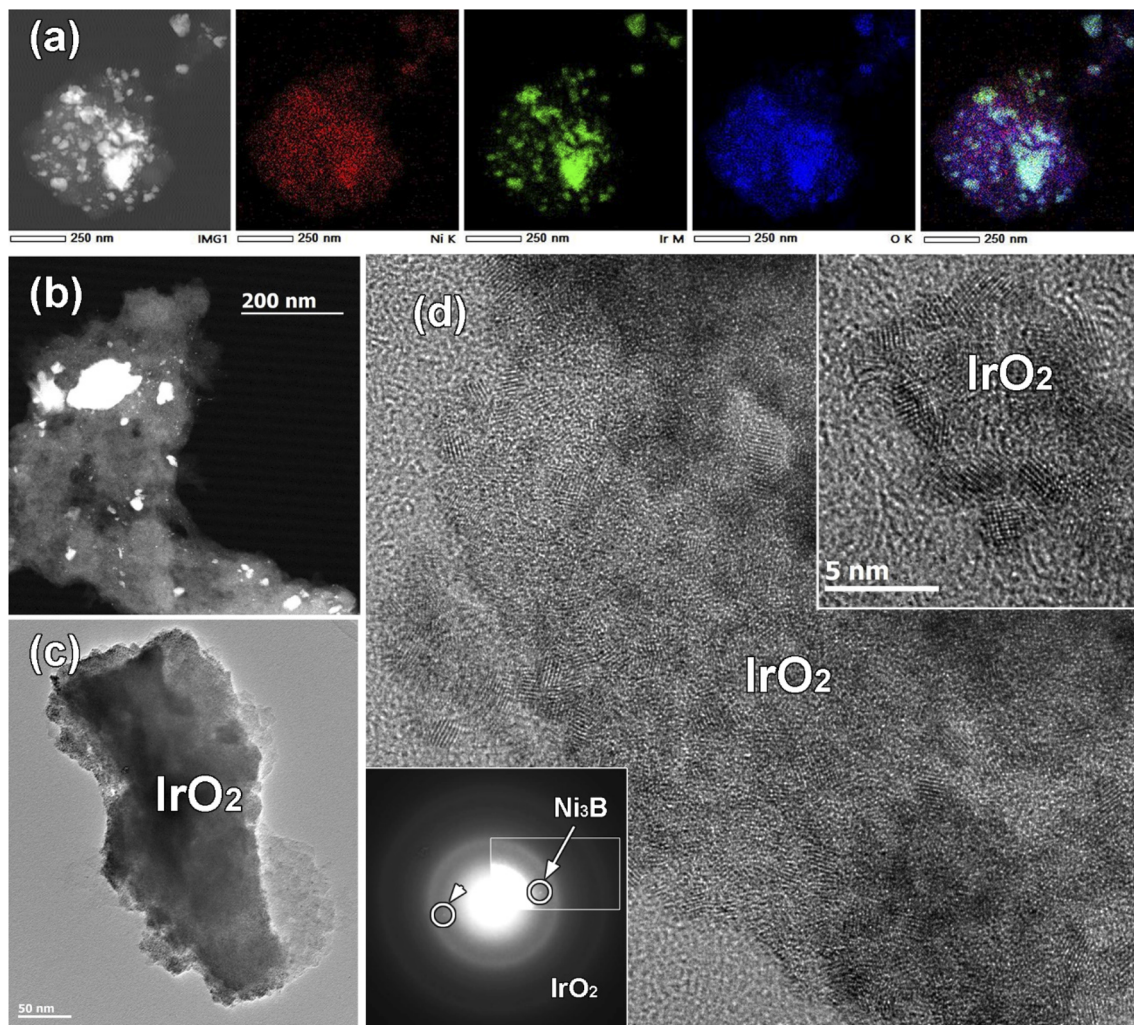


Fig. 7 TEM results of  $\text{Ni}_3\text{B}/\text{IrO}_2$  sample after the catalysis test. (a) Low magnification HAADF-STEM image and simultaneously acquired EDX-STEM elemental mapping for Ni-K, Ir-M, O-K edges and overlaid color images. (b) Low magnification HAADF-STEM image. (c) Low magnification TEM image of  $\text{IrO}_2$  nanoparticles agglomerated. (d) Bright field HRTEM image of agglomeration of  $\text{IrO}_2$  nanoparticles. Top right inset: cluster of  $\text{IrO}_2$  nanoparticles. Bottom left inset: selected area electron diffraction (SAED) pattern with highlighted diffraction spots of  $\text{Ni}_3\text{B}$ .

molar ratio can achieve a similar “apparent” stability as pure  $\text{IrO}_2$ , while simultaneously maintaining most of the OER activity. Better anchoring of  $\text{IrO}_2$  into the support may improve the overall catalyst’s stability. For further improvement, particle size control or nanostructuring of metal borides can be achieved by taking advantage of the topotactic nature of the proposed synthetic method, improving the surface interaction with  $\text{IrO}_2$ .<sup>31,32,45,46</sup> In addition, probing different metal borides and ternary mixed metal borides with better intrinsic stability might be valuable. For both purposes, the herein reported method is a viable synthetic avenue.

## Conclusion

A facile and prompt low-temperature method of synthesizing various metal borides spanning across the transition metal group was developed. The borodization was achieved by

reaction of a metal source with  $\text{BI}_3$ , producing the metal boride and a washable metal iodide admixture. The method was shown to be able to produce mixed metal borides as well as to topotactically transform complex metal morphologies, such as Ni foam, into corresponding metal boride morphologies, *i.e.*,  $\text{Ni}_3\text{B}$  foam. *In situ* studies of the reaction mechanism revealed that the formation of the metal iodide byproduct is a sufficient driving force to initiate metal boride formation at temperatures as low as 700 K. From an applications point-of-view, we have shown the potential of  $\text{Ni}_3\text{B}$  to become a viable stable supporting material for the state-of-the-art  $\text{IrO}_2$  electrocatalyst for OER in acidic conditions. The composite electrocatalyst with 50%  $\text{Ni}_3\text{B}$  and 50%  $\text{IrO}_2$  exhibits similar activity and stability in acidic OER as pure  $\text{IrO}_2$  electrocatalyst. Further investigations on using metal borides alone as OER electrocatalysts under acidic conditions, or the addition of metal borides for promotion of OER activity while simultaneously reducing the usage of



expensive platinum group metal electrocatalysts, will be of great interest.

## Data availability

The data that support the findings of this study are available from the corresponding authors upon reasonable request.

## Author contributions

The manuscript was written through contributions of all authors.

## Conflicts of interest

Authors declare no conflict of interests.

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