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## Coupling of PET waste electroreforming with green hydrogen generation using bifunctional catalyst†

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Cost-effective and high-efficiency bifunctional electrocatalysts for electrooxidation of polyethylene terephthalate (PET) waste and green hydrogen generation are very crucial for practical implementation yet rarely reported. Herein, a bifunctional catalyst of cobalt modified nickel phosphide nanosheet arrays on nickel foam (Co-Ni<sub>2</sub>P/NF) for both PET hydrolysate oxidation reaction and hydrogen evolution reaction (HER) is reported, which is obtained by a facile hydrothermal and phosphidation treatment. The electrocatalyst is highly active for both PET hydrolysate oxidation reaction and HER with low overpotentials of 90 and 148 mV, respectively, to achieve a current density of 50 mA cm<sup>-2</sup>. By coupling PET hydrolysate oxidation reaction with HER, the assembled electrolyzer with Co-Ni<sub>2</sub>P/NF as a bifunctional catalyst only requires 1.43 V to afford 10 mA cm<sup>-2</sup>, much lower than that needed for pure water splitting (1.55 V). Complementary DFT study provides an in-depth understanding of HER and electrooxidation of PET on Co-Ni<sub>2</sub>P/NF. Our work suggests that electroreforming of abundant PET waste could be an energy-efficient and sustainable strategy for both plastic waste valorization and green hydrogen production *via* using a cost-effective and active bifunctional catalyst.

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## Introduction

Hydrogen (H<sub>2</sub>) has been dubbed “the cleanest energy source in the world” because its combustion product is pure water. Moreover, H<sub>2</sub> is also a stable media for storing intermittent renewable energy, thus is expected to play a pivotal role for a sustainable energy future.<sup>1–3</sup> Nevertheless, only green H<sub>2</sub> is clean and emits negligible carbon in its life cycle. Water splitting is the only method to produce green H<sub>2</sub>.<sup>4–6</sup> A thermodynamic voltage of at least 1.23 V is required for H<sub>2</sub>O molecular breakage, where substantial amount of energy is used to oxidize water that produces the by-product oxygen.<sup>7,8</sup> Therefore, replacing water oxidation with a new oxidation reaction to reduce energy consumption and to produce higher value product (than oxygen) is attracting ever-increasing interest.<sup>9,10</sup> Electrooxidation of many different organic molecules has been reported to couple with HER for green H<sub>2</sub> generation, such as biomass,<sup>11–13</sup> hydrazine,<sup>14,15</sup> isopropanol,<sup>16</sup> benzylamine,<sup>17</sup> urea,<sup>18</sup> formic acid,<sup>19</sup> and so on.

To this end, polyethylene terephthalate (PET) plastic, as an organic polymer, also can be used to assist the production of green H<sub>2</sub>. Due to its excellent tensile strength, thermal stability, electrical insulation, processability, chemical resistance, non-toxicity, and tastelessness, PET is often used in electronic devices, clothing, and product packaging, *etc.*<sup>20</sup> Among them, PET bottles account for about one third of the total amount of PET produced (~24 million tons),<sup>21</sup> and are widely used for food, chemical, and pharmaceutical packaging. Although PET bottles have brought a lot of convenience to our daily life, the natural degradation of their waste requires hundreds of years, which causes serious environmental issues, *i.e.*, “white pollution”.<sup>22</sup> The recycling of PET waste can not only address environment issue but also recover important resources, being very crucial for sustainable development. Therefore, it is undoubtedly attractive to combine the electroreforming of waste PET bottles with H<sub>2</sub> production.

As far as we know, only few reports to date focus on the plastic electroreforming-assisted green H<sub>2</sub> generation. Duan and coworkers successfully synthesized CoNi<sub>0.25</sub>P electrocatalyst for electrocatalytic upcycling of PET to produce formate and H<sub>2</sub>. The results reveal that the Faradaic efficiency and selectivity of formate reach more than 80% in a membrane-electrode assembly reactor when the current density is 500 mA cm<sup>-2</sup>.<sup>21</sup> Zhao's group successfully converted PET to formate by using CuO NWs/CF and meanwhile assisted H<sub>2</sub> production. Experimental and density functional theory (DFT) calculation demonstrated that the optimal reaction pathway on CuO NWs is

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to generate the glyoxal intermediate and obtain formate through the C-C bond cleavage with a high selectivity of 86.5%.<sup>23</sup> However, pure PET plastics were used in these studies, and there is no report on the electroreforming of raw plastic waste coupled with H<sub>2</sub> generation yet.

Herein, we electroreform raw waste PET bottles to assist H<sub>2</sub> production *via* electrolysis. We firstly explore the maximum solubility of waste PET bottles in sodium hydroxide (NaOH), and then optimize an efficient and active electrocatalyst for electrooxidation of PET bottles hydrolysate, namely cobalt-modified nickel phosphide nanosheets on the nickel foam (Co-Ni<sub>2</sub>P/NF). Moreover, Co-Ni<sub>2</sub>P/NF serving as a bifunctional electrocatalyst are assembled to form an electrolytic cell, which only requires a voltage of 1.43 V to achieve a current density of 10 mA cm<sup>-2</sup> in 1 M NaOH electrolyte with 5 g L<sup>-1</sup> PET, and the Faradaic efficiency of formate can reach 85%, while the recovery of terephthalic acid (TPA) was as high as 80%.

## Experimental section

### The synthesis of Co-modified Ni<sub>2</sub>P nanosheets on Ni foam (Co-Ni<sub>2</sub>P/NF)

To produce a series of ultrathin Co<sub>x</sub>-Ni<sub>2</sub>P/NF nanosheets ( $x = 0.2, 0.4, 0.6, 0.8$ ), (1 -  $x$ ) mL 0.5 M nickel(II) chloride hexahydrate (NiCl<sub>2</sub>·6H<sub>2</sub>O),  $x$  mL 0.5 M cobaltous chloride (CoCl<sub>2</sub>·6H<sub>2</sub>O), and 1 mL ammonium hydroxide (NH<sub>3</sub>·H<sub>2</sub>O) were mixed to obtain a uniform solution. And then 1 × 1 cm<sup>2</sup> nickel foam (NF) was immersed into the uniform solution and 1 mL hydrazine hydrate (N<sub>2</sub>H<sub>4</sub>) solution was added. The solution was then heated at 75 °C for 10 h, obtaining layered cobalt modified nickel hydroxide precursor on NF [termed as Co<sub>x</sub>-Ni(OH)<sub>2</sub>/NF].

Afterward, a piece of Co<sub>x</sub>-Ni(OH)<sub>2</sub>/NF and 0.05 g sodium hypophosphite (NaH<sub>2</sub>PO<sub>2</sub>·H<sub>2</sub>O) were placed in two different combustion boats with NaH<sub>2</sub>PO<sub>2</sub>·H<sub>2</sub>O boat located at upstream of a tube furnace and then heated at 300 °C for 2 h with a flow of argon. The excess PH<sub>3</sub> in the product was absorbed by the saturated copper(II) sulfate (CuSO<sub>4</sub>) solution after treatment at the downstream. Finally, the product (Co<sub>x</sub>-Ni<sub>2</sub>P/NF) was obtained after being naturally cooled down to room temperature, followed by repeat water cleaning, and then drying at 60 °C for 10 h.

CoP/NF and Ni<sub>2</sub>P/NF were also fabricated as control samples. The synthetic method of CoP/NF nanosheets and Ni<sub>2</sub>P/NF nanosheets was similar to that of Co<sub>x</sub>-Ni<sub>2</sub>P/NF nanosheets, except that the mixture solution of NiCl<sub>2</sub>·6H<sub>2</sub>O and CoCl<sub>2</sub>·6H<sub>2</sub>O was substituted by pure CoCl<sub>2</sub>·6H<sub>2</sub>O or NiCl<sub>2</sub>·6H<sub>2</sub>O, respectively.

### Electrochemical characterizations

HER characterization and PET hydrolysate electrooxidation process were carried out in a typical three-electrode system. Electrocatalyst supported on NF was used as working electrode, Ag/AgCl electrode and carbon rod acted as reference electrode and counter electrode, respectively. LSV curves were compensated according to:  $E_{\text{compensated}} = E_{\text{measured}} - IR_s$ , where  $R_s$  is the solution resistance extracted from the electrochemical

impedance spectroscopy (EIS) measurement.<sup>24</sup> And according to the equation  $E_{\text{RHE}} = E_{\text{Ag/AgCl}} + 0.197 \text{ V} + 0.0591 \text{ pH}$ , all electrode potentials have been calibrated as reversible hydrogen electrode (RHE).

In addition, the Faradaic efficiency (FE) for the PET electrooxidation to formate is determined by equation:  $\text{FE} = (3Fn)/Q$ , where  $F$  is Faraday's constant (96 485 C mol<sup>-1</sup>),  $n$  represents the moles of the formate produced, and  $Q$  is the total charge passed across the electrode during electrolysis.<sup>21,25</sup>

### Hydrolysis of waste PET bottles

The mixture of various white waste PET plastic bottles was briefly rinsed with water and dried at 70 °C (for precise weighing). Different amounts of the PET bottle fragment (0.5 g, 1.25 g, 3.75 g) were added to 25 mL 10 M sodium hydroxide (NaOH) solution, and stirred at 80 °C for 10 h. After cooling to room temperature, water was added to increase solution volume to 250 ml, and then it was stirred until the PET fragments were completely dissolved. The electrolytes with different PET contents (marked as 2 g L<sup>-1</sup>, 5 g L<sup>-1</sup>, 15 g L<sup>-1</sup>) were prepared.

### Computational method

All the calculations were carried out using the density functional theory (DFT) with the generalized Perdew-Burke-Ernzerhof (PBE) with DFT +  $U$  ( $U_{\text{eff}} = U - J = 5.3$  for Ni and  $U_{\text{eff}} = U - J = 4.0$  for Co) and the projector augmented-wave (PAW) pseudopotential plane-wave method<sup>27</sup> as implemented in the VASP code.<sup>28</sup> For the PAW pseudopotential, we included 3d<sup>8</sup>4s<sup>1</sup>, 3d<sup>8</sup>4s<sup>2</sup>, 2s<sup>2</sup>2p<sup>4</sup>, 2s<sup>2</sup>2p<sup>2</sup> and 1s<sup>1</sup> were treated as valence electrons for Co, Ni, O, C and H atoms, respectively. A 5 × 5 × 3 Monkhorst-Pack (MP)  $k$ -point grid was used for Ni<sub>2</sub>P bulk optimization and a plane-wave basis set with an energy cutoff of 400 eV considering the spin polarization. Good convergence was obtained with these parameters and the total energy was converged to 1.0 × 10<sup>-6</sup> eV per atom. The optimized bulk model was used to build the (201) surface unitcell and further expanded to a 6-atom layer 3 × 2 × 1 supercell with a vacuum separation of ~18 Å for H adsorption and ethylene glycol (EG) reduction process studies. We carried out calculations with the van der Waals (vdW) correction by employing optB86b-vdW functional using a 2 × 2 × 1 MP  $k$ -point grid.<sup>29</sup>

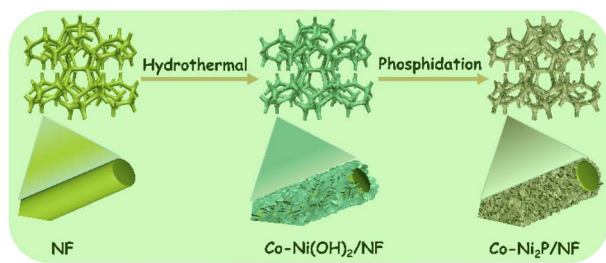
## Results and discussion

### Synthesis and characterization of electrocatalysts

Co-Ni<sub>2</sub>P/NF electrode was successfully obtained through a facile two-step process. As shown in the Scheme 1, cobalt modified nickel hydroxide nanosheet arrays on nickel foam (denoted as Co-Ni(OH)<sub>2</sub>/NF) was first achieved (Fig. S1†). Secondly, the final electrocatalyst of Co-Ni<sub>2</sub>P/NF was obtained by phosphidation with NaH<sub>2</sub>PO<sub>2</sub>·H<sub>2</sub>O at 300 °C for 2 h.

The morphological characterizations of Co-Ni<sub>2</sub>P/NF were analyzed by the scanning electron microscopy (SEM). As depicted in Fig. S2,† the morphologies of four Co-Ni<sub>2</sub>P/NF samples are nanosheets, indicating that the variation of Co content does not affect the unique lamellar structure of the





Scheme 1 Schematic of the synthetic method of Co-Ni<sub>2</sub>P/NF.

electrocatalysts. We chose Co<sub>0.6</sub>-Ni<sub>2</sub>P/NF for more detailed characterization. As shown in Fig. 1A and B, the Co<sub>0.6</sub>-Ni<sub>2</sub>P/NF nanosheets are uniformly distributed over the entire surface of 3D porous NF and inherit the highly cross-linked ultrathin layered structure of Co-Ni(OH)<sub>2</sub>. The chemical composition of the Co<sub>0.6</sub>-Ni<sub>2</sub>P/NF were characterized by SEM-mapping, energy-dispersive X-ray spectrum (EDS), and the X-ray diffraction (XRD). The mapping images display the uniform distribution of Ni, Co, and P element (Fig. 1C), indicating the successful synthesis of metal phosphide. And the EDS spectrum further proves the Co element in the product (Fig. 1D). As depicted in Fig. 1E, the XRD pattern suggests that the remaining peaks perfect match with (111), (210), (300) and (211) planes of hexagonal Ni<sub>2</sub>P (JCPDS no. 74-1385) expect for the NF substrate (JCPDS no. 04-0850), suggesting the main component of the product is Ni<sub>2</sub>P. Due to its low content and the influence of the substrate, the peaks of cobalt species could not be observed. Putting together, the Co modified Ni<sub>2</sub>P nanosheets were successfully obtained.

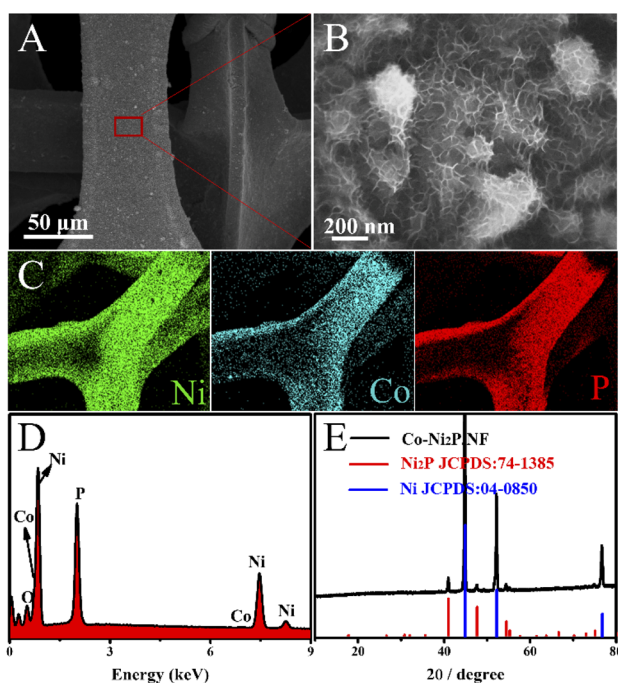


Fig. 1 (A) and (B) SEM images, (C) SEM-mapping, (D) SEM-EDS, and (E) XRD pattern of Co<sub>0.6</sub>-Ni<sub>2</sub>P/NF.

Transmission electron microscopy (TEM) images further show that Co modified Ni<sub>2</sub>P nanosheets have been synthesized successfully (Fig. 2A). More detail about the composition and structures were established by high-resolution transmission electron microscopy (HRTEM). As shown in Fig. 2B, the HRTEM reveals that the heterointerface structure exists in the ultrathin Co<sub>0.6</sub>-Ni<sub>2</sub>P nanosheets, and the interfaces between CoP and Ni<sub>2</sub>P marked with yellow dotted line can be clearly observed. The lattice fringes in different regions can match with two different material of CoP and Ni<sub>2</sub>P, respectively. The interplanar spacing of 0.203 nm is agreement with the (201) plane of Ni<sub>2</sub>P, and the lattice fringe spacing of 0.189 nm can be indexed to (211) plane of CoP. The EDS elemental mapping analysis of Co<sub>0.6</sub>-Ni<sub>2</sub>P nanosheets are depicted in Fig. 2C. It can be observed that the elements of Co, Ni and P are uniformly dispersed in the nanosheets. Furthermore, the composition ratio of CoP and Ni<sub>2</sub>P in Co<sub>0.6</sub>-Ni<sub>2</sub>P/NF is about 1 : 4 (Fig. 2D), as obtained from EDS analysis.

As presented in Fig. 3, the chemical valence of Co<sub>0.6</sub>-Ni<sub>2</sub>P/NF was analyzed by X-ray photoelectron spectroscopy (XPS). The Co, Ni and P elements can be obviously seen from the full spectrum of Co<sub>0.6</sub>-Ni<sub>2</sub>P/NF (Fig. 3A). As illustrated in Fig. 3B, two narrow peaks at 874.5 eV and 856.7 eV in the Ni 2p XPS spectrum correspond to Ni 2p<sub>1/2</sub> and Ni 2p<sub>3/2</sub> of the oxidized Ni species, while the peaks located at 861.8 eV and 880.3 eV belong to satellite peaks.<sup>30,31</sup> Furthermore, other two peaks centered at 852.9 eV and 869.8 eV are consistent with Ni-P bonds.<sup>32-34</sup> On the Co 2p XPS spectrum, the two peaks located at 778.6 eV and 793.7 eV correspond to Co 2p<sub>3/2</sub> and Co 2p<sub>1/2</sub> of Co-P, respectively. And the two peaks located at 782.0 eV and 798.2 eV are consistent with Co-O bond, with two satellites detected at 803.4 eV and 786.0 eV (Fig. 3C).<sup>35,36</sup> As presented in Fig. 3D, one can observe two peaks located at 129.0 eV and 130.0 eV, which are corresponding to P 2p<sub>3/2</sub> and P 2p<sub>1/2</sub>, respectively.<sup>37</sup> In

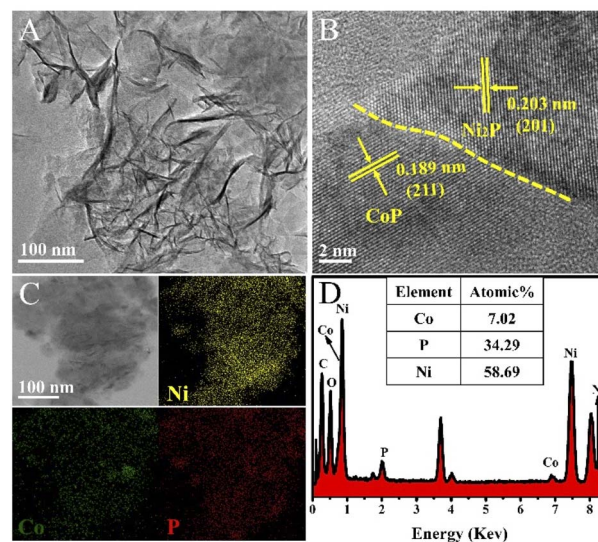


Fig. 2 (A) TEM image, (B) HRTEM image, (C) TEM-mapping, and (D) TEM-EDS analysis of Co<sub>0.6</sub>-Ni<sub>2</sub>P/NF.



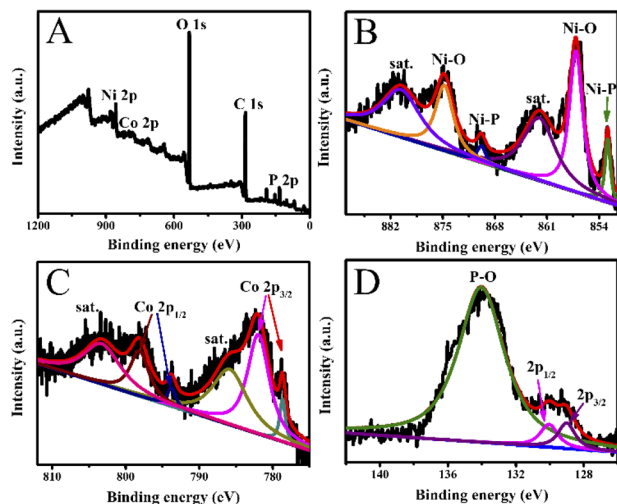


Fig. 3 (A) XPS survey spectrum, and high-resolution XPS spectra of (B) Ni 2p, (C) Co 2p, and (D) P 2p of  $\text{Co}_{0.6}\text{-Ni}_2\text{P/NF}$ .

addition, a broad peak located at 134.0 eV is originating from a surface oxidation.<sup>38–40</sup>

### HER electrocatalytic performance of the prepared catalysts

As shown in Fig. 4A, HER activity was assessed by linear sweep voltammetry (LSV) curves in 1 M NaOH electrolyte.  $\text{Co}_{0.6}\text{-Ni}_2\text{P/NF}$  exhibits the highest HER activity, which only requires an overpotential of 69 mV to achieve a current density of  $10 \text{ mA cm}^{-2}$ . As depicted in Fig. 4B and D, the overpotential (148 mV)

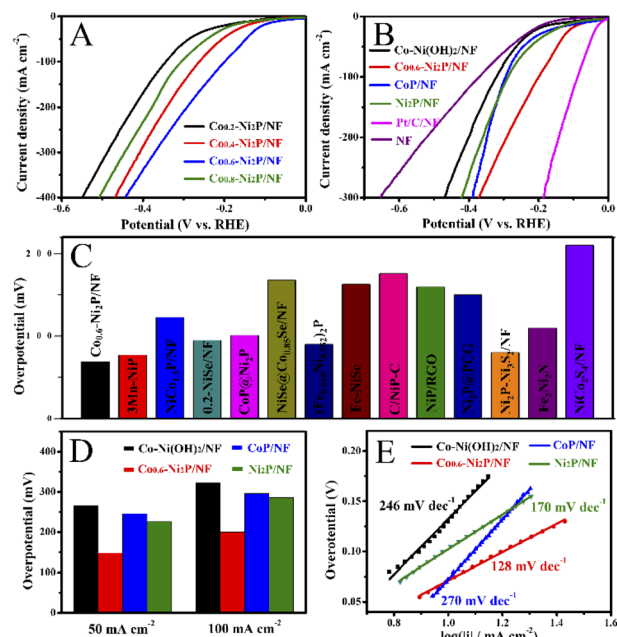


Fig. 4 LSV curves of (A)  $\text{Co-Ni}_2\text{P/NF}$  with different Co contents in a 1 M NaOH solution and (B)  $\text{Co-Ni(OH)}_2\text{/NF}$ ,  $\text{CoP/NF}$ ,  $\text{Ni}_2\text{P/NF}$ ,  $\text{Co}_{0.6}\text{-Ni}_2\text{P/NF}$  and  $\text{Pt/C/NF}$ . Comparison of the overpotential at current densities of (C)  $10 \text{ mA cm}^{-2}$  and (D) 50 and  $100 \text{ mA cm}^{-2}$ . (E) Tafel plots of  $\text{Co-Ni(OH)}_2\text{/NF}$ ,  $\text{CoP/NF}$ ,  $\text{Ni}_2\text{P/NF}$ , and  $\text{Co}_{0.6}\text{-Ni}_2\text{P/NF}$ .

of  $\text{Co}_{0.6}\text{-Ni}_2\text{P/NF}$  at  $50 \text{ mA cm}^{-2}$  is much smaller than that of  $\text{Ni}_2\text{P/NF}$  (226 mV),  $\text{CoP/NF}$  (246 mV),  $\text{Co-Ni(OH)}_2\text{/NF}$  (266 mV) and  $\text{NF}$  (284 mV). To achieve a current density of  $100 \text{ mA cm}^{-2}$ ,  $\text{Co}_{0.6}\text{-Ni}_2\text{P/NF}$  only requires an overpotential of 200 mV (Fig. 4D). Notably, although the HER activity of  $\text{Co}_{0.6}\text{-Ni}_2\text{P/NF}$  is inferior to that of  $\text{Pt/C/NF}$  (24 mV), it is still superior to most of reported Ni/Co-based HER electrocatalysts (Fig. 4C).<sup>14,41–52</sup>

As illustrated in Fig. 4E, the Tafel slope value of  $\text{Co}_{0.6}\text{-Ni}_2\text{P/NF}$ ,  $\text{Ni}_2\text{P/NF}$ ,  $\text{CoP/NF}$ , and  $\text{Co-Ni(OH)}_2\text{/NF}$  are 128, 170, 297, and  $275 \text{ mV dec}^{-1}$ , respectively, suggesting that  $\text{Co}_{0.6}\text{-Ni}_2\text{P/NF}$  exhibits significantly faster HER kinetics than other electrocatalysts. As presented in Fig. 5A, the electrochemical impedance spectroscopy (EIS) analysis shows that  $\text{Co}_{0.6}\text{-Ni}_2\text{P/NF}$  has the lowest  $R_{\text{ct}}$  value ( $7.6 \Omega$ ) among  $\text{Co-Ni(OH)}_2\text{/NF}$  ( $46.4 \Omega$ ),  $\text{CoP/NF}$  ( $8.5 \Omega$ ), and  $\text{Ni}_2\text{P/NF}$  ( $10.8 \Omega$ ), suggesting  $\text{Co}_{0.6}\text{-Ni}_2\text{P/NF}$ 's strongest charge transfer ability and fastest kinetics among all electrocatalysts tested.<sup>53,54</sup> The electrochemically active surface (ECSA) of the electrocatalysts are obtained by calculating the double layer capacitances ( $C_{\text{dl}}$ ) (Fig. S3†). As depicted in Fig. 5B,  $\text{Co}_{0.6}\text{-Ni}_2\text{P/NF}$  exhibits a highest  $C_{\text{dl}}$  ( $20.1 \text{ mF cm}^{-2}$ ) and thus largest ECSA among  $\text{Co-Ni(OH)}_2\text{/NF}$  ( $11.7 \text{ mF cm}^{-2}$ ),  $\text{CoP/NF}$  ( $19.2 \text{ mF cm}^{-2}$ ), and  $\text{Ni}_2\text{P/NF}$  ( $19.9 \text{ mF cm}^{-2}$ ). The improved ECSA can be ascribed to the rough surface and 3D porous nanostructure that offers abundant active sites. Undoubtedly, the increase of the active edges/sites is favorable to enhance the interface with electrolyte, the adsorption of water molecule, and the release of  $\text{H}_2$ , thereby enhancing the HER electrocatalytic activity.<sup>55,56</sup> Fig. 5C shows that  $\text{Co}_{0.6}\text{-Ni}_2\text{P/NF}$  exhibits higher intrinsic HER activity (normalized by ECSA), implying that the high total activity of  $\text{Co}_{0.6}\text{-Ni}_2\text{P/NF}$  is not only due to its large ECSA but also to its high intrinsic activity.

We further support the excellent HER performance of  $\text{Co-Ni}_2\text{P/NF}$  with theoretical investigation. The optimized unit cell

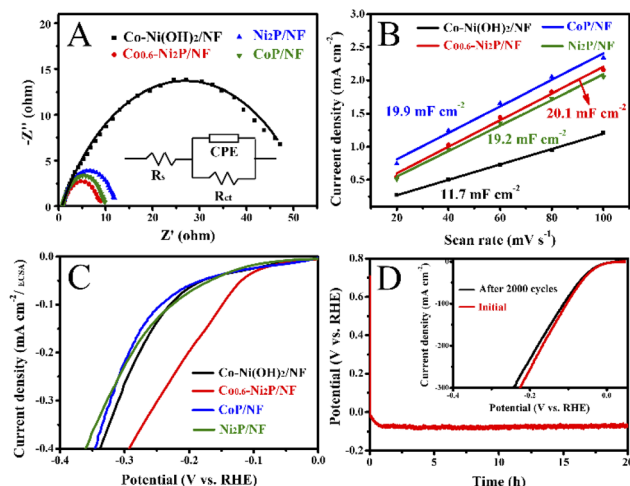


Fig. 5 (A) Nyquist plots acquired in 1 M NaOH electrolyte at  $-0.2 \text{ V vs. RHE}$  with the corresponding fitting curves and the equivalent circuit (inset). (B) Plots of the current density at  $0.32 \text{ V vs. scan rate}$ . (C) ECSA-normalized LSV curves of  $\text{Co-Ni(OH)}_2\text{/NF}$ ,  $\text{CoP/NF}$ ,  $\text{Ni}_2\text{P/NF}$ , and  $\text{Co}_{0.6}\text{-Ni}_2\text{P/NF}$ . (D) Chronopotentiometry curve of  $\text{Co}_{0.6}\text{-Ni}_2\text{P/NF}$  in 1 M NaOH solution at constant current density of  $10 \text{ mA cm}^{-2}$  (inset: polarization curves of  $\text{Co}_{0.6}\text{-Ni}_2\text{P/NF}$  before and after 2000 cycles).



of Ni<sub>2</sub>P was shown in Fig. S4.† The expanded (210) supercell contains 108 Ni and 54 P atoms. Only one Co atom was considered embedded into the Ni<sub>2</sub>P surface. Considering three different embedding sites for Co on the surface and the relative energy reveal that the Co prefer to replace Ni at the 4-fold site.

The first step of HER is the electrochemical adsorption process:  $H^+ + e^- \rightarrow H^*$ , the second step is the electrochemical discharge process:  $H^* + H^+ + e^- \rightarrow H_2$  or chemical desorption process:  $2H^* \rightarrow H_2$ .<sup>57</sup> The Gibbs free energy of  $\Delta G_H$  has been successfully used as a suitable descriptor for evaluating the HER catalytic activity of various materials. Generally, the optimal value of  $\Delta G_H$  is 0 eV, where hydrogen is adsorbed neither too strongly nor too weakly to facilitate both hydrogen adsorption and desorption.<sup>57</sup>

The adsorption energy ( $\Delta E_H$ ) is computed as

$$\Delta E_H = E(*H) - E(*) - 1/2 E(H_2)$$

where  $E(*H)$  and  $E(*)$  are the total energy of a surface with and without hydrogen adsorption, respectively, and  $E(H_2)$  is the total energy of a H<sub>2</sub> molecule.

The Gibbs free energy of H ( $\Delta G_H$ ) is defined as:

$$\Delta G_H = \Delta E_H + \Delta E_{ZPE} - T\Delta S_H$$

where  $\Delta E_H$  is the adsorption energy,  $\Delta E_{ZPE}$  is the difference in zero-point energy,  $T$  is the temperature (298 K) and  $\Delta S_H$  is the entropy difference between H that is adsorbed and in the gas phase. We approximated the entropy of hydrogen adsorption as  $\Delta S_H \approx 1/2 (S_{H_2}^\circ)$ , where  $S_{H_2}^\circ$  is the entropy of gas phase H<sub>2</sub> at standard conditions.

The calculated Gibbs free energies of hydrogen ( $\Delta G_H$ ) on the pristine and Co doped Ni<sub>2</sub>P surfaces are shown in Fig. S5.† It can be observed that the  $\Delta G_H$  is 418 meV on the pristine surface and dramatically decreases to 190 meV on Co doped surface. Based on our simulation results, it can be considered that doping Co atoms in Ni<sub>2</sub>P can reduce the HER overpotential. It should be noted that only one Co atom was embedded into Ni<sub>2</sub>P systems. Since the active site for HER is changed to Co, we consider that more Co doping within the solubility may further reduce the overpotential. Notably, our theoretical results are consistent with the experimental values, in which it was found that Co<sub>0.6</sub>-Ni<sub>2</sub>P/NF exhibits the highest HER activity with an overpotential of 69 mV.

Finally, the stability of Co<sub>0.6</sub>-Ni<sub>2</sub>P/NF in 1 M NaOH is accessed by the chronopotentiometric curve. As depicted in Fig. 5D, there was no noticeable decrease of HER current density during a 20 h continuous electrolysis. Indeed, SEM image reveals the morphology of Co<sub>0.6</sub>-Ni<sub>2</sub>P/NF retained well after 20 h continuous operation thanks to the great structural integrity due to the direct growth of Co<sub>0.6</sub>-Ni<sub>2</sub>P on NF (Fig. S6A†). Moreover, the HER activity was retained well with a small increase of only 10 mV at 100 mA cm<sup>-2</sup> after 2000 potential cycles (inset of Fig. 5D). It should be noted that the composition of Co<sub>0.6</sub>-Ni<sub>2</sub>P/NF after HER test showed no noticeable changes (Fig. S6B-F†). The excellent HER performances of Co<sub>0.6</sub>-Ni<sub>2</sub>P/NF can be explained as follows. Firstly, the

*in situ* growth of Co<sub>0.6</sub>-Ni<sub>2</sub>P on NF support not only reinforces the interaction between Co<sub>0.6</sub>-Ni<sub>2</sub>P nanosheet and NF, enhancing the structural integrity, but also improves electrical conductivity, facilitating efficient charge transport.<sup>58-60</sup> Secondly, Co<sub>0.6</sub>-Ni<sub>2</sub>P possesses ultrathin nanosheet structure and a large electrochemical active area, beneficial for the exposure of more active sites. Also, the 3D porous structure can provide abundant channels to facilitate ion transfer and release of product (H<sub>2</sub>), thus giving rise to the boosted catalytic kinetics.<sup>61,62</sup> Thirdly, the phosphorus and metal in Co<sub>0.6</sub>-Ni<sub>2</sub>P nanosheets play an important role in capturing nearby protons and act as acceptor centers for adsorbed hydrides, promoting HER process.<sup>63,64</sup> Lastly, Co doping and the interface formed between CoP and Ni<sub>2</sub>P can further improve the catalytic activity by adjusting the electronic structure of the electrocatalyst.<sup>65-67</sup>

### The effect of PET hydrolysate concentration on electrooxidation activity

Fig. 6A and B show that Co<sub>0.6</sub>-Ni<sub>2</sub>P/NF nanosheet exhibits the highest current density for oxygen evolution reaction (OER) at the same voltage in 1 M NaOH. The LSV curves show that the voltage required for the PET hydrolysate electrooxidation of Co<sub>0.6</sub>-Ni<sub>2</sub>P/NF is significantly lower than that of OER at the same current density (Fig. 6C). To achieve 100 mA cm<sup>-2</sup>, Co<sub>0.6</sub>-Ni<sub>2</sub>P/NF only requires an overpotential of 174 mV in 1 M NaOH with 2 g L<sup>-1</sup> PET, which is much lower than that of OER (350 mV). CV curves further show that the PET hydrolysate electrooxidation current density of Co<sub>0.6</sub>-Ni<sub>2</sub>P/NF at 1.5 V vs. RHE has reached 183 mA cm<sup>-2</sup> while that of OER is only 46 mA cm<sup>-2</sup> (Fig. 6D). These observations indicate that PET hydrolysate electrooxidation is much more favorable than OER thermodynamically.

Fig. 7A shows that the current density gradually increases with the addition of 0.5 mL PET hydrolysate (5 g L<sup>-1</sup>) at 1.5 V.

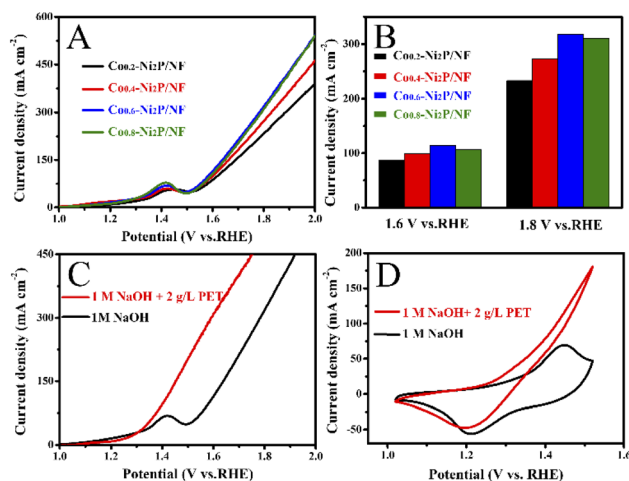


Fig. 6 (A) LSV curves of Co-Ni<sub>2</sub>P/NF with different Co contents in a 1 M NaOH solution. (B) Comparison of the current densities at potential of 1.6 V and 1.8 V vs. RHE. (C) LSV curves of Co<sub>0.6</sub>-Ni<sub>2</sub>P/NF in 1 M NaOH with or without 2 g L<sup>-1</sup> PET. (D) CV curves of Co<sub>0.6</sub>-Ni<sub>2</sub>P/NF in 1 M NaOH with or without 2 g L<sup>-1</sup> PET.



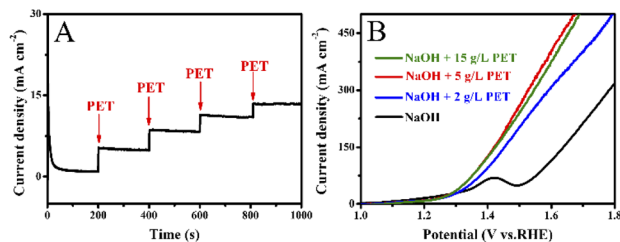


Fig. 7 (A) Chronoamperometry curves of  $\text{Co}_{0.6}\text{-Ni}_2\text{P/NF}$  at 1.5 V vs. RHE after adding PET hydrolysate into 1 M NaOH solution. (B) LSV curves of  $\text{Co}_{0.6}\text{-Ni}_2\text{P/NF}$  in 1 M NaOH with different PET hydrolysate content.

However, the magnitude of each increase in current density gradually decreases. As illustrated in Fig. 7B, one can observe that the current density increases at first and then decreases with increasing PET hydrolysate concentration, and peaks at 5 g  $\text{L}^{-1}$ . Too high concentration ( $>5$  g  $\text{L}^{-1}$ ) could severely affect the mass transfer and ionic conductivity, and thus decrease the total activity.

### PET hydrolysate electrooxidation activity comparison

Next, we investigate the PET hydrolysate electrooxidation performance of the five electrocatalysts under the same experimental conditions. As depicted in Fig. 8A, the PET hydrolysate electrooxidation activity of  $\text{Co-Ni}_2\text{P/NF}$  is much better than other electrocatalysts. Specifically,  $\text{Co-Ni}_2\text{P/NF}$  only requires an overpotential of 90 mV to reach a current density of 50  $\text{mA cm}^{-2}$ , which is much smaller than those of  $\text{CoP/NF}$  (140 mV),

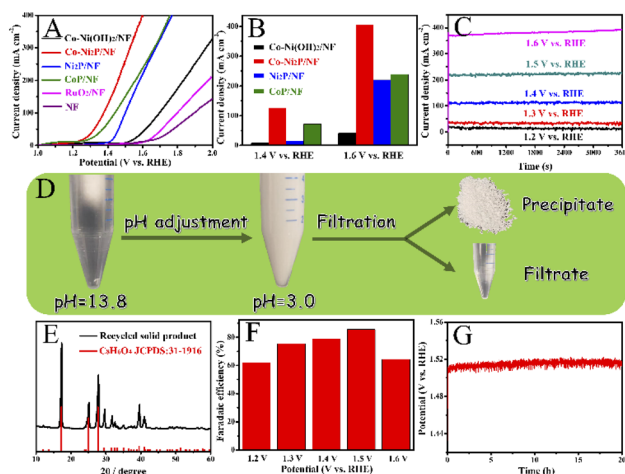


Fig. 8 (A) LSV curves of  $\text{Co-Ni}_2\text{P/NF}$ ,  $\text{Ni}_2\text{P/NF}$ ,  $\text{CoP/NF}$ ,  $\text{Co-Ni(OH)}_2/\text{NF}$ ,  $\text{RuO}_2/\text{NF}$  and  $\text{NF}$  anodes in the electrolyte of 1 M NaOH with 5 g  $\text{L}^{-1}$  PET. (B) Comparison of the current density of  $\text{Co-Ni}_2\text{P/NF}$ ,  $\text{Ni}_2\text{P/NF}$ ,  $\text{CoP/NF}$  and  $\text{Co-Ni}_2(\text{OH})_2/\text{NF}$  at 1.4 V vs. RHE and 1.6 V vs. RHE. (C) Chronoamperometry curves of  $\text{Co-Ni}_2\text{P/NF}$  at the corresponding potentials. (D) Pictures for separation of TPA and formic acid. (E) The XRD pattern of the recycled solid product with standard TPA spectrum. (F) The Faradaic efficiencies of PET oxidation to formate at each given potential of  $\text{Co-Ni}_2\text{P/NF}$ . (G) Continuous electrolysis using  $\text{Co-Ni}_2\text{P/NF}$  over 20 h in 1 M NaOH electrolyte with 5 g  $\text{L}^{-1}$  PET at current density of 200  $\text{mA cm}^{-2}$ .

$\text{Ni}_2\text{P/NF}$  (230 mV),  $\text{Co-Ni(OH)}_2/\text{NF}$  (385 mV),  $\text{RuO}_2/\text{NF}$  (422 mV) and  $\text{NF}$  (544 mV). The current density of  $\text{Co-Ni}_2\text{P/NF}$  at 1.4 V and 1.6 V are 125 and 402  $\text{mA cm}^{-2}$ , respectively, which are higher than those of the rest electrocatalysts (Fig. 8B). Continuous electrolysis for 1 h at different voltages (Fig. 8C) was conducted, followed by pH adjust using HCl. White substances were precipitated during this process (Fig. 8D). The liquid electrolyte and precipitate were separated by filtration, wherein the precipitate was confirmed to be terephthalic acid (TPA,  $\text{C}_8\text{H}_6\text{O}_4$ ) by XRD with recovery yield of 80% (Fig. 8E). Additionally, the liquid product was characterized by high-performance liquid chromatography (HPLC). As depicted in Fig. S7,† the content of ethylene glycol (EG) was the highest at 1.2 V vs. RHE, and the electrooxidation rate of EG accelerated when the applied voltage increased. After 1 h electrolysis, the main product was formate, and its maximum FE was about 85% at 1.5 V vs. RHE (Fig. 8F).

To further verify the durability of  $\text{Co-Ni}_2\text{P/NF}$  for electro-reforming of PET hydrolysate, continuous electrolysis was conducted for 20 h at a constant current density of 200  $\text{mA cm}^{-2}$ , as illustrated in Fig. 8G. The imperceptible potential change shows the excellent stability of  $\text{Co-Ni}_2\text{P/NF}$  electrode for PET electroreforming. Moreover, the morphology of  $\text{Co-Ni}_2\text{P/NF}$  was dominated by nanosheets after 20 h continuous electrolysis (Fig. S8A†). As depicted in Fig. S8B–F,†  $\text{Co-Ni}_2\text{P/NF}$  underwent partial reconstruction and oxidation after PET hydrolysate oxidation, generating metal oxy(hydroxide) analogue. Based on our experimental results and previous studies,<sup>21,25</sup> we proposed the process of waste PET bottle recycling, as shown in Fig. S9,† which mainly includes three steps, namely: (1) waste PET bottle hydrolysis (red part), (2) electrooxidation of EG (blue part) and (3) the separation of TPA and formic acid (black part).

We have also investigated the energy landscape of electro-oxidation ethylene glycol (EG). Here, we proposed the possible reaction steps and the sketch of the electrooxidation process of the EG is shown in Fig. S10.†

The electrooxidation reaction energy ( $\Delta E_{\text{EG}}^i$ ) of each step is defined as



where \* represents the adsorption site and the adsorption energy of EG on the surface was considered the reference. It should be noted we only considered the reaction energy ignoring the vibration contribution to the reaction process.

The calculated energy landscapes of EG electrooxidation on the pristine and Co doped  $\text{Ni}_2\text{P}$  surfaces are shown in Fig. S11.† We calculated the reaction energy of each step based on the proposed EG electrooxidation process. In comparison, removing two protons from EG ( $\Delta E_{\text{EG}}^1$ ) needs 2.193 eV on the pristine  $\text{Ni}_2\text{P}$  surface, it dramatically decreases to 1.243 eV on Co doped  $\text{Ni}_2\text{P}$  surface. However, the value of  $\Delta E_{\text{EG}}^2$  is 1.916 eV,



slightly lower than 2.043 eV with Co doping. Therefore, we consider that embedding Co atoms can't reduce the energy step for removing two additional protons. Breaking the C–C bond needs the energy of 3.400 eV on the pristine Ni<sub>2</sub>P surface, which is much higher than that on Co doped Ni<sub>2</sub>P surface of 1.774 eV. The final step of adsorption of OH<sup>−</sup> is an exothermic reaction on the pristine and Co doped Ni<sub>2</sub>P surface. Overall, breaking the C–C bonds is the energy-determining step for the electro-oxidation of EG based on our DFT simulations, and chemical doping of Co should be responsible for enhancing the reaction process and reduce the energy consumption for EG electro-oxidation reactions.

### PET hydrolysate electroreforming-assisted water splitting

Lastly, we explore the waste PET bottle electroreforming-assisted water splitting in a two-electrode system with Co-Ni<sub>2</sub>P/NF as the bifunctional electrode in 1 M NaOH electrolyte with 5 g L<sup>−1</sup> PET. As depicted in Fig. 9A, the Co-Ni<sub>2</sub>P/NF||Co-Ni<sub>2</sub>P/NF electrolyzer requires an overall voltage of 1.43 V to achieve 10 mA cm<sup>−2</sup>, which is much less than that of the pure water electrolysis (1.55 V), indicating the PET electrooxidation-assisted water electrolyzer has higher energy conversion efficiency for H<sub>2</sub> fuel than traditional water electrolyzer. During the electrolysis process, a great deal of H<sub>2</sub> gas bubbles was observed from the cathode and the value-added chemicals (sodium formate and disodium terephthalate) were obtained from the anode (Fig. 9B), which effectively eliminated the potential explosion risk of H<sub>2</sub>/O<sub>2</sub> mixture in the traditional water electrolyzer. Additionally, the potential of the Co-Ni<sub>2</sub>P/NF||Co-Ni<sub>2</sub>P/NF electrolyzer (1.62 V) was much lower than that of commercial electrocatalysts, *i.e.*, Pt/C/NF||RuO<sub>2</sub>/NF electrolyzer (1.92 V) to achieve current density of 50 mA cm<sup>−2</sup> (Fig. 9C). Furthermore, chronopotentiometric curves display that Co-Ni<sub>2</sub>P/NF||Co-Ni<sub>2</sub>P/NF

NF electrolyzer is more stable (with unnoticeable voltage change) than the Pt/C/NF||RuO<sub>2</sub>/NF electrolyzer (voltage increase of 40 mV) during 10 h continuous electrolysis (Fig. 9D).

## Conclusion

We have successfully developed cobalt modified nickel phosphide electrocatalyst supported on Ni foam (Co-Ni<sub>2</sub>P/NF) by a facile hydrothermal and phosphidation treatment. With a large specific surface area, eminent electrical conductivity, and abundant active sites, the resultant Co-Ni<sub>2</sub>P/NF catalyst acts as a highly active and robust bifunctional electrocatalyst for HER and PET electrooxidation. The Co element is atomically dispersed in the Ni<sub>2</sub>P structure, forming unsaturated atomic heterometallic Co<sub>x</sub>Ni<sub>1−x</sub>P sites that can effectively regulate the electronic structure of Ni and P atoms. As a result, the catalyst exhibits superior activity for electrooxidation PET hydrolysate in alkaline media, affording 50 mA cm<sup>−2</sup> at a low potential of 1.32 V. A two-electrode electrolyzer with the Co-Ni<sub>2</sub>P/NF as a bifunctional electrocatalyst can achieve 10 mA cm<sup>−2</sup> current density at a low voltage of 1.43 V for concurrent PET electroreforming and green H<sub>2</sub> generation. This work not only provides a new facile method for the synthesis of bifunctional electrocatalysts through surface and interface engineering, but also opens a pathway for a sustainable recycling of PET bottles.

## Author contributions

Ying Li and Li Quan Lee: methodology, investigation, characterization, and writing—original draft. Hu Zhao: investigation and characterization. Zhi Gen Yu and Yong-Wei Zhang: density functional theory (DFT) calculation. Hong Li and Pingqi Gao: supervision, validation, writing—review & editing, and funding acquisition. All authors have given approval to the final version of the manuscript.

## Conflicts of interest

There are no conflicts to declare.

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## References

- 1 T. N. Veziroğlu and S. Şahin, *Energy Convers. Manage.*, 2008, **49**, 1820–1831.
- 2 H. Balat and E. Kirtay, *Int. J. Hydrogen Energy*, 2010, **35**, 7416–7426.

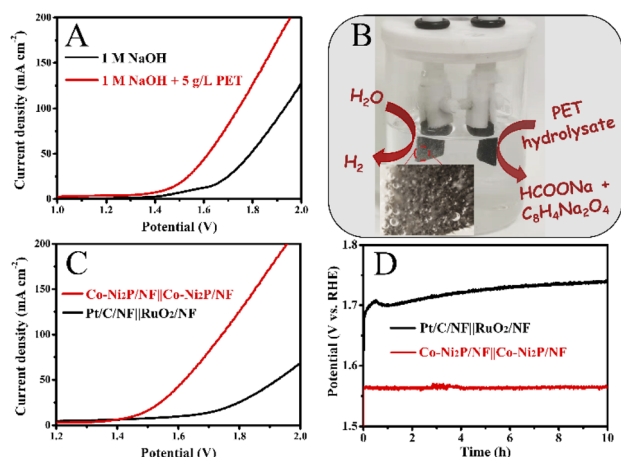


Fig. 9 (A) Polarization curves of Co-Ni<sub>2</sub>P/NF||Co-Ni<sub>2</sub>P/NF electrolyzer in 1 M NaOH with or without 5 g L<sup>−1</sup> PET, (B) photograph of the Co-Ni<sub>2</sub>P/NF||Co-Ni<sub>2</sub>P/NF electrolyzer during electrolysis process. (C) Polarization curves of the Co-Ni<sub>2</sub>P/NF||Co-Ni<sub>2</sub>P/NF and Pt/C/NF||RuO<sub>2</sub>/NF electrolyzers in 1 M NaOH with 5 g L<sup>−1</sup> PET, (D) chronopotentiometric curves of the Co-Ni<sub>2</sub>P/NF||Co-Ni<sub>2</sub>P/NF and Pt/C/NF||RuO<sub>2</sub>/NF electrolyzers at 10 mA cm<sup>−2</sup>.



- 3 M. Li, Z. Xu, Y. Li, J. Wang and Q. Zhong, *Nano Select*, 2021, **3**, 147–156.
- 4 N. Grimaldos-Osorio, F. Sordello, M. Passananti, P. Vernoux and A. Caravaca, *J. Power Sources*, 2020, **480**, 228800.
- 5 S. Shiva Kumar and V. Himabindu, *Mater. Sci. Energy Technol.*, 2019, **2**, 442–454.
- 6 H. Li, H. Li, Y. Qiu, S. Liu, J. Fan and X. Guo, *Nano Select*, 2021, **2**, 2148–2158.
- 7 A. Buttler and H. Spliethoff, *Renewable Sustainable Energy Rev.*, 2018, **82**, 2440–2454.
- 8 H. Okada, E. Tsuji, M. Hisada, S. Kitano, H. Habazaki, S. Suganuma and N. Katada, *Sustainable Energy Fuels*, 2022, **6**, 2709–2717.
- 9 K. Ye, G. Wang, D. Cao and G. Wang, *Top. Curr. Chem.*, 2018, **376**, 42.
- 10 Y. Li, Z. Dang and P. Gao, *Nano Select*, 2021, **2**, 847–864.
- 11 G. Yang, Y. Jiao, H. Yan, Y. Xie, A. Wu, X. Dong, D. Guo, C. Tian and H. Fu, *Adv. Mater.*, 2020, **32**, 2000455.
- 12 H. Zhao, D. Lu, J. Wang, W. Tu, D. Wu, S. W. Koh, P. Gao, Z. J. Xu, S. Deng, Y. Zhou, B. You and H. Li, *Nat. Commun.*, 2021, **12**, 2008.
- 13 S. Sreekantan, A. A. Balachandran Kirali and B. Marimuthu, *Sustainable Energy Fuels*, 2022, **6**, 3681–3689.
- 14 Y. Li, Y. Zhao, F. M. Li, Z. Dang and P. Gao, *ACS Appl. Mater. Interfaces*, 2021, **13**, 34457–34467.
- 15 Q. Sun, Y. Li, J. Wang, B. Cao, Y. Yu, C. Zhou, G. Zhang, Z. Wang and C. Zhao, *J. Mater. Chem. A*, 2020, **8**, 21084–21093.
- 16 Y. Zhao, S. Xing, X. Meng, J. Zeng, S. Yin, X. Li and Y. Chen, *Nanoscale*, 2019, **11**, 9319–9326.
- 17 Y. Ding, B.-Q. Miao, S.-N. Li, Y.-C. Jiang, Y.-Y. Liu, H.-C. Yao and Y. Chen, *Appl. Catal., B*, 2020, **268**, 118393.
- 18 M. Yao, J. Ge, B. Sun, J. Hu, S. W. Koh, Y. Zhao, J. Fei, Z. Sun, W. Hong, Z. Chen, W. Hu and H. Li, *Chem. Eng. J.*, 2021, **414**, 128753.
- 19 Q. Xue, X. Y. Bai, Y. Zhao, Y. N. Li, T. J. Wang, H. Y. Sun, F. M. Li, P. Chen, P. Jin, S. B. Yin and Y. Chen, *J. Energy Chem.*, 2022, **65**, 94–102.
- 20 L. Wang, G. A. Nelson, J. Toland and J. D. Holbrey, *ACS Sustainable Chem. Eng.*, 2020, **8**, 13362–13368.
- 21 H. Zhou, Y. Ren, Z. Li, M. Xu, Y. Wang, R. Ge, X. Kong, L. Zheng and H. Duan, *Nat. Commun.*, 2021, **12**, 4679.
- 22 X. Jiao, K. Zheng, Q. Chen, X. Li, Y. Li, W. Shao, J. Xu, J. Zhu, Y. Pan, Y. Sun and Y. Xie, *Angew. Chem., Int. Ed.*, 2020, **59**, 1–6.
- 23 J. Wang, X. Li, T. Zhang, Y. Chen, T. Wang and Y. Zhao, *J. Phys. Chem. Lett.*, 2022, **13**, 622–627.
- 24 H. Wang, Y. Li, R. Wang, B. He and Y. Gong, *Electrochim. Acta*, 2018, **284**, 504–512.
- 25 J. Wang, X. Li, M. Wang, T. Zhang, X. Chai, J. Lu, T. Wang, Y. Zhao and D. Ma, *ACS Catal.*, 2022, **12**, 6722–6728.
- 26 J. P. Perdew, K. Burke and M. Ernzerhof, *Phys. Rev. Lett.*, 1996, **77**, 3865–3868.
- 27 P. E. Blöchl, *Phys. Rev. B*, 1994, **50**, 17953–17979.
- 28 G. Kresse and J. Furthmüller, *Comput. Mater. Sci.*, 1996, **6**, 15–50.
- 29 J. Klimeš, D. R. Bowler and A. Michaelides, *Phys. Rev. B*, 2011, **83**, 195131.
- 30 J. Liu, C. Li, Q. Ye, L. Lin, Y. Wang, M. Sun and Y. Cheng, *Catal. Commun.*, 2022, **163**, 106416.
- 31 S. Li, Y. Zou, L. Li, Y. Dong and Y. Zhu, *ChemistrySelect*, 2019, **4**, 6295–6303.
- 32 T. Wang, X. Cao and L. Jiao, *eScience*, 2021, **1**, 69–74.
- 33 J. Xu, Y. Zhang, Z. Liu, Y. H. Chen and Z. Liu, *Mater. Today Commun.*, 2021, **29**, 102731.
- 34 Y. Liu, B. Wang, K. Srinivas, M. Wang, Z. Chen, Z. Su, D. Liu, Y. Li, S. Wang and Y. Chen, *Int. J. Hydrogen Energy*, 2022, **47**, 12903–12913.
- 35 Y. Li, F. Li, Y. Zhao, S. N. Li, J. H. Zeng, H. C. Yao and Y. Chen, *J. Mater. Chem. A*, 2019, **7**, 20658–20666.
- 36 J. Sun, M. Ren, L. Yu, Z. Yang, L. Xie, F. Tian, Y. Yu, Z. Ren, S. Chen and H. Zhou, *Small*, 2019, **15**, 1804272.
- 37 X. Yang, A. Y. Lu, Y. Zhu, M. N. Hedhili, S. Min, K. W. Huang, Y. Han and L. J. Li, *Nano Energy*, 2015, **15**, 634–641.
- 38 L. Ji, J. Wang, X. Teng, T. J. Meyer and Z. Chen, *ACS Catal.*, 2019, **10**, 412–419.
- 39 Y. Li, H. Zhang, M. Jiang, Y. Kuang, X. Sun and X. Duan, *Nano Res.*, 2016, **9**, 2251–2259.
- 40 T. J. Wang, Y. C. Jiang, J. W. He, F. M. Li, Y. Ding, P. Chen and Y. Chen, *Carbon Energy*, 2022, **4**, 283–293.
- 41 C. X. Zhang, D. Li and Y. Xu, *J. Mater. Res.*, 2022, **37**, 807–817.
- 42 T. Kong, Y. W. Sui, J. Q. Qi, F. X. Wei, Y. J. Ren, Z. Z. Zhan, Z. Sun, M. H. Zhou, D. M. Meng, L. J. Zhang, L. Ma and Q. Wang, *J. Mater. Sci.: Mater. Electron.*, 2021, **32**, 1951–1961.
- 43 H. Jin, S. Liu, L. Pei, G. Li, Z. F. Ma, W. F. Bai, S. T. Wu, Y. J. Yuan and J. S. Zhong, *RSC Adv.*, 2021, **11**, 22467–22472.
- 44 W. L. Ding, Y. H. Cao, H. Liu, A. X. Wang, C. J. Zhang and X. R. Zheng, *Rare Met.*, 2021, **40**, 1373–1382.
- 45 W. Zhang, Y. Zou, H. Liu, S. Chen, X. Wang, H. Zhang, X. She and D. Yang, *Nano Energy*, 2019, **56**, 813–822.
- 46 Z. Zou, X. Wang, J. Huang, Z. Wu and F. Gao, *J. Mater. Chem. A*, 2019, **7**, 2233–2241.
- 47 F. Wang, Y. M. Liu and C. Y. Zhang, *New J. Chem.*, 2019, **43**, 4160–4167.
- 48 Z. Li, J. Shang, C. Su, S. Zhang, M. Wu and Y. Guo, *J. Fuel Chem. Technol.*, 2018, **46**, 473–478.
- 49 M. Miao, R. Hou, Z. Liang, R. Qi, T. He, Y. Yan, K. Qi, H. Liu, G. Feng and B. Y. Xia, *J. Mater. Chem. A*, 2018, **6**, 24107–24113.
- 50 L. Zeng, K. Sun, X. Wang, Y. Liu, Y. Pan, Z. Liu, D. Cao, Y. Song, S. Liu and C. Liu, *Nano Energy*, 2018, **51**, 26–36.
- 51 M. Jiang, Y. Li, Z. Lu, X. Sun and X. Duan, *Inorg. Chem. Front.*, 2016, **3**, 630–634.
- 52 A. Sivanantham, P. Ganesan and S. Shanmugam, *Adv. Funct. Mater.*, 2016, **26**, 4661–4672.
- 53 S. Wang, J. Li, R. Zhang and L. Li, *Mater. Lett.*, 2017, **198**, 61–64.
- 54 X. H. Ji, Z. M. Wang, X. Y. Chen and R. B. Yu, *Chem. J. Chin. Univ.*, 2021, **42**, 1377–1394.
- 55 S. Huo, S. Yang, Q. Niu, F. Yang and L. Song, *Int. J. Hydrogen Energy*, 2020, **45**, 4015–4025.



- 56 X. Jin, J. Li, Y. Cui, X. Liu, K. Wang, Y. Zhou, W. Yang, X. Zhang, C. Zhang, X. Jiang and B. Liu, *Int. J. Hydrogen Energy*, 2019, **44**, 5739–5747.
- 57 B. Hinnemann, P. G. Moses, J. Bonde, K. P. Jørgensen, J. H. Nielsen, S. Hørch, I. Chorkendorff and J. K. Nørskov, *J. Am. Chem. Soc.*, 2005, **127**, 5308–5309.
- 58 S. Yang, J. Y. Zhu, X. N. Chen, M. J. Huang, S. H. Cai, J. Y. Han and J. S. Li, *Appl. Catal., B*, 2022, **304**, 120914.
- 59 H. Sun, Z. Yan, F. Liu, W. Xu, F. Cheng and J. Chen, *Adv. Mater.*, 2020, **32**, 1806326.
- 60 Z. B. Li, J. Wang, X. J. Liu, R. Li, H. Wang, Y. Wu, X. Z. Wang and Z. P. Lu, *Scr. Mater.*, 2019, **173**, 51–55.
- 61 J. Liu, D. Zhu, Y. Zheng, A. Vasileff and S.-Z. Qiao, *ACS Catal.*, 2018, **8**, 6707–6732.
- 62 M. Guo, Y. Qu, F. Zeng and C. Yuan, *Electrochim. Acta*, 2018, **292**, 88–97.
- 63 J. K. Das, A. K. Samantara, S. Satyarthi, C. S. Rout and J. N. Behera, *RSC Adv.*, 2020, **10**, 4650–4656.
- 64 R. Boppella, J. Tan, W. Yang and J. Moon, *Adv. Funct. Mater.*, 2018, **29**, 1807976.
- 65 W. Li, D. Xiong, X. Gao, W.-G. Song, F. Xia and L. Liu, *Catal. Today*, 2017, **287**, 122–129.
- 66 Y. Ji, J. Xie, Y. Yang, X. Fu, R. Sun and C. Wong, *Chin. Chem. Lett.*, 2020, **31**, 855–858.
- 67 C. Du, L. Yang, F. Yang, G. Cheng and W. Luo, *ACS Catal.*, 2017, **7**, 4131–4137.

