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## Bicarbonate insertion triggered self-assembly of chiral octa-gold nanoclusters into helical superstructures in the crystalline state†

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Constructing atomically precise helical superstructures of high order is an extensively pursued subject for unique aesthetic features and underlying applications. However, the construction of cluster-based helices of well-defined architectures comes with a huge challenge owing to their intrinsic complexity in geometric structures and synthetic processes. Herein, we report a pair of unique *P*- and *M*-single stranded helical superstructures spontaneously assembled from *R*- and *S*-Au8c individual nanoclusters, respectively, upon selecting chiral BINAP (2,2'-bis(diphenylphosphino)-1,1'-binaphthalene) and hydrophilic *o*-H<sub>2</sub>MBA (*o*-mercaptopbenzoic acid) as protective ligands to induce chirality and facilitate the formation of helices. Structural analysis reveals that the chirality of the Au8c individual nanoclusters is derived from the homochiral ligands and the inherently chiral Au<sub>8</sub> metallic kernel, which was further corroborated by experimental and computational investigations. More importantly, driven by the O–H···O interactions between (HCO<sub>3</sub><sup>–</sup>)<sub>2</sub> dimers and achiral *o*-HMBA<sup>–</sup> ligands, *R/S*-Au8c individual nanoclusters can assemble into helical superstructures in a highly ordered crystal packing. Electrospray ionization (ESI) and collision-induced dissociation (CID) mass spectrometry of Au8c confirm the hydrogen-bonded dimer of Au8c individual nanoclusters in solution, illustrating that the insertion of (HCO<sub>3</sub><sup>–</sup>)<sub>2</sub> dimers plays a crucial role in the assembly of helical superstructures in the crystalline state. This work not only demonstrates an effective strategy to construct cluster-based helical superstructures at the atomic level, but also provides visual and reliable experimental evidence for understanding the formation mechanism of helical superstructures.

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## Introduction

Chirality appears to be widespread in nature and occurs on various length scales ranging from the subatomic to molecular, to supramolecular level, and even to galaxies.<sup>1–3</sup> In contrast to molecular chirality that essentially refers to point, axis and plane chirality, supramolecular chirality is represented by the

periodic asymmetric packing of chiral or achiral molecules, which is ubiquitous in biological systems.<sup>4–7</sup> A typical example of supramolecular chirality is the secondary structural proteins which display miscellaneous architectures, such as the  $\alpha$ -helix,  $\beta$ -sheet, and coiled-coil helix bundle. Within the subset of chiral superstructures, the helix is a typical topological structure that can either be left-handed (*P*) or right-handed (*M*) depending on the screwing motion directions.<sup>8,9</sup> In recent years, helical superstructures have received considerable attention due to their fascinating geometric features and promising applications in the fields of chiral recognition, sensors and asymmetric catalysis.<sup>10–12</sup> However, how to synthesize helical superstructures with precise size, structure and composition is still a problem to be solved.

Supramolecular interactions (non-covalent interactions), such as hydrogen bonding,  $\pi\cdots\pi$  stacking, C–H··· $\pi$ , and electrostatic interaction, are the driving forces for constructing and crystallizing aesthetical supramolecular architectures with high regularity at the atomic level.<sup>13–16</sup> Through these supramolecular interactions, many supramolecular architectures based on coinage metal nanoclusters of precise structures and compositions have been successfully constructed.<sup>17–22</sup> As we know, some

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supramolecular helical structures related to hydrogen bond interactions can be established *via* introducing carboxylic acids. However, to date, cluster-based helical architectures have been rarely reported due to the lack of structural stability and difficulty in atomic-level characterization.<sup>23–27</sup> Thus, in-depth investigation of helical superstructures of ordered arrangement units at the atomic level is still in the hysteretic state.

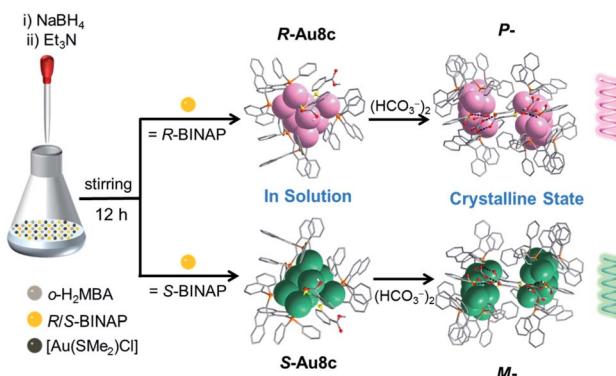
Protective ligands on the surface of coinage metal nanoclusters are particularly crucial for dominating their sizes, structural patterns and physicochemical properties; moreover, appropriate choice of surface ligands is directly correlated to their packing in the crystalline state,<sup>28–34</sup> which may result in diverse superstructures. Since the observation of circular dichroism (CD) signals in ligand-protected gold nanoclusters by Schaff and Whetten in 1998,<sup>35</sup> a series of chiral organic ligands have been extensively employed to functionalize coinage metal nanoclusters and significant progress has been achieved in recent years.<sup>36–44</sup> Besides, it is found that protective ligands containing –COOH groups are of potential interest for constructing various cluster-based supramolecular architectures *via* intermolecular interactions. For example, both the structurally determined  $\text{Ag}_{44}$  and  $\text{Au}_{102}$  nanoclusters protected by the *p*-mercaptopbenzoic acid (*p*-H<sub>2</sub>MBA) ligands displayed well-organized arrangements in their crystal lattices driven by hydrogen bonding.<sup>17,45–47</sup> Thus, the selection of suitable surface ligands determines not only the structures and properties of the metal nanoclusters but also the superstructures formed upon entering condensed states.

Here, we exquisitely select chiral 2,2'-bis(diphenylphosphino)-1,1'-binaphthalene (BINAP) as the major protective ligand to endow gold nanoclusters with chirality, and *o*-mercaptopbenzoic acid (*o*-H<sub>2</sub>MBA) carrying –COOH groups as the auxiliary ligand to crystallize diverse gold cluster-based superstructures. A pair of enantiomers  $[\text{Au}_8(\text{R/S-BINAP})_3(\text{o-HMBA})_2] \cdot 2\text{HCO}_3^-$  (**R/S-Au8c**) in which two  $\text{HCO}_3^-$  anions were formed *in situ* by capturing the atmospheric  $\text{CO}_2$  has been successfully synthesized. Strikingly, **R-Au8c** and **S-Au8c** demonstrate *P*- and *M*-helical superstructures running along the 2-fold screw axis in the crystalline state, respectively, driven by the synergistic effect of intermolecular hydrogen bonds and electrostatic interactions. Their chirality was elucidated *via* X-ray crystallography, CD spectra and theoretical calculations, which indicated that the chirality arose from the metallic kernel and ligand shell. It was also found that **Au8c** exhibits crystallization-induced emission enhancement (CIEE).<sup>48</sup> This study provides an excellent structural model to understand the self-assembly of cluster-based supramolecular helices at the atomic level and inspires us to construct helical superstructures in a controllable manner.

## Results and discussion

### Synthesis discussion

In a representative reaction, a mixture of  $[\text{Au}(\text{SMe}_2)\text{Cl}]$ , *R*- or *S*-BINAP and *o*-H<sub>2</sub>MBA was dissolved in a mixed solvent of  $\text{CH}_3\text{OH}$  and  $\text{CH}_2\text{Cl}_2$ . A freshly prepared aqueous solution of  $\text{NaBH}_4$  was employed to initiate the reduction process, followed by the addition of  $\text{Et}_3\text{N}$  at room temperature (Scheme 1). After two



Scheme 1 Synthetic routes to **R/S-Au8c** in solution and the homochiral *P*-/*M*-helices formed in the crystalline state.

weeks, plate-like orange crystals of **Au8c** were obtained by layering *n*-hexane on the dichloromethane solutions of the corresponding gold nanoclusters, with a yield of  $\sim 15\%$  based on  $[\text{Au}(\text{SMe}_2)\text{Cl}]$  (Fig. S1†). Single-crystal X-ray diffraction (SCXRD) analysis revealed that the unit cell of **Au8c** contains two unexpected  $\text{HCO}_3^-$  anions, which were connected together as a  $(\text{HCO}_3^-)_2$  dimer by hydrogen bonds, without an additional source in the reaction. Under alkaline conditions, such  $\text{HCO}_3^-$  or carbonates ( $\text{CO}_3^{2-}$ ) were occasionally observed in coinage metal nanoclusters and they are usually derived from the conversion of atmospheric  $\text{CO}_2$ .<sup>49–51</sup> Thus we speculated that the  $\text{HCO}_3^-$  anions were generated *in situ* by capturing atmospheric  $\text{CO}_2$  in the presence of  $\text{Et}_3\text{N}$  during the reaction process ( $\text{CO}_2 + \text{H}_2\text{O} + \text{Et}_3\text{N} \rightarrow \text{Et}_3\text{NH}^+ + \text{HCO}_3^-$ ).<sup>52</sup>

To prove our hypothesis, we performed a controlled experiment under a  $\text{CO}_2$  atmosphere represented by **R-Au8c**. As speculated, the yield ( $\sim 25\%$ ) of **R-Au8c** nanoclusters was slightly increased. For comparison, we also performed an experiment under a  $\text{N}_2$  atmosphere, and another new nanocluster,  $[\text{Au}_8(\text{R-BINAP})_3(\text{o-MBA})_2]$  (**R-Au8d**), was isolated without any trapped  $\text{HCO}_3^-$  anions (Fig. S2 and Table S1†). Furthermore, different amounts of  $\text{NaHCO}_3/\text{KHCO}_3$  ranging from 0.005 mmol to 0.03 mmol were directly added to the reaction solution to improve the yield of **R-Au8c**, but no crystalline products were obtained. Based on the above observation, intentionally added  $\text{HCO}_3^-$  may disrupt the acid–base equilibrium of the reaction system ( $\text{OH}^- + \text{HCO}_3^- \rightleftharpoons \text{H}_2\text{O} + \text{CO}_3^{2-}$ ), which was unfavourable for the crystallization of **R-Au8c**.

Moreover, we also tried to replace *o*-H<sub>2</sub>MBA with *m*-H<sub>2</sub>MBA and *p*-H<sub>2</sub>MBA ligands to explore the effect of positional isomers on the helical arrangement of  $\text{Au}_8$  nanoclusters. When *o*-H<sub>2</sub>MBA was replaced by *m*-H<sub>2</sub>MBA,  $[\text{Au}_8(\text{R-BINAP})_3(\text{m-MBA})_2]$  (**R-Au8e**) was obtained (Fig. S3 and Table S1†); whereas when *o*-H<sub>2</sub>MBA was replaced by *p*-H<sub>2</sub>MBA, no crystalline products were obtained. These results suggested that even a subtle variation of positional isomeric ligands (*o*/*m*/*p*-H<sub>2</sub>MBA) can make a dramatic effect on the formation and helical arrangement of nanoclusters (for some detailed discussions please see the below section). The solid-state Fourier transform infrared (FT-IR) spectrum further confirmed the presence of  $\text{HCO}_3^-$  anions in **Au8c**, and it was comparable to that of  $\text{NaHCO}_3$  (Fig. S4†) and

similar to that in the literature.<sup>53</sup> Phase purity of **R**-/S-**Au8c** was characterized by powder X-ray diffraction (PXRD) (Fig. S5†). More details of synthesis and characterization are shown in the ESI.†

### Crystal structures of **R**/S-**Au8c**

SCXRD analysis revealed that the basic unit of **R**/S-**Au8c** contains one  $[\text{Au}_8(\text{R/S-BINAP})_3(\text{o-HMBA})_2]^{2+}$  nanocluster and two  $\text{HCO}_3^-$  anions. Both **R-Au8c** and **S-Au8c** crystallize in the orthorhombic chiral space group  $P2_12_12_1$ , with Flack parameters of 0.012(12) and 0.041(8), respectively, indicating their absolutely chiral configurations (Table S1†). To better understand their geometric structures, the dicationic **R**/S-**Au8c** enantiomers without anions are depicted in Fig. 1. The **Au8c** individual nanocluster has an **Au<sub>7</sub>** core, which can be regarded as the fusion of an **Au<sub>5</sub>** trigonal bipyramidal and an **Au<sub>5</sub>** square pyramid through sharing a triangular **Au<sub>3</sub>** face, where the **Au<sub>7</sub>** core is attached by an exterior gold atom in **R-Au8c** and **S-Au8c**, respectively (Fig. S6†). Such an additional gold atom at the asymmetric *exo* position lowers the symmetry of **Au8c**, leading to the  $C_1$  symmetry and perfect mirror symmetry of the **Au<sub>8</sub>** metallic kernels. Thus, **Au8c** demonstrates an intrinsically chiral kernel, which is not usual in chiral gold nanoclusters.<sup>36,41,42</sup> The average Au–Au bond length of **R-Au8c** is 2.83 Å (range: 2.65–3.21 Å), which is slightly shorter than the 2.88 Å in bulk gold, indicating the existence of aurophilic interactions.<sup>54</sup>

Compared to the lowest-energy planar **Au<sub>8</sub>** isomers derived from DFT calculations,<sup>55</sup> **Au8c** exhibits a new type of nonplanar configuration that can be ascribed to the surface ligand effects.<sup>30</sup> On the periphery of **Au8c**, there are totally three  $\mu_2$ -bridging chiral BINAP and two  $\mu_2$ -bridging  $\text{o-HMBA}^-$  ligands. Each BINAP ligand has a flexible conformation due to the rotation of the C–C single bond, which is unconducive to the planar geometry of **Au<sub>8</sub>**; all of them ride on the edges of the **Au<sub>8</sub>** metallic kernel to form contorted  $\text{Au}_2\text{P}_2\text{C}_4$  eight-membered rings (average Au–P: 2.29 Å in **R-Au8c**). One of the two  $\text{o-HMBA}^-$  ligands bridge the exterior gold atom (Au1) and one Au atom (Au6) at the vertex of the shared **Au<sub>3</sub>** triangle, and the other bridges a base edge (Au4–Au7) of the **Au<sub>5</sub>** square pyramid by

a  $\mu_2$ -S atom (Fig. S6†). The bond angles of Au1–S1–Au4 and Au7–S2–Au8 in **R-Au8c** are 97.3(2)° and 92.1(3)° (Table S2†), respectively, which are smaller than the typical Au–S–Au average bond angle in thiolate-protected gold nanoclusters.<sup>56,57</sup>

Unexpected Au···H–C electrostatic interactions between the gold framework and the chiral BINAP in **R**/S-**Au8c**, which are a new kind of hydrogen bond according to IUPAC definitions in 2011,<sup>58</sup> are highlighted. In terms of the axially chiral BINAP ligands, two biaryl backbones can agilely rotate around the axis to form a dihedral angle, further making the aromatic rings inclined to establish Au···H–C interactions. In the crystal structures of **R-Au8c** and **S-Au8c**, the hydrogen atoms from the phenyl rings of BINAP are close to the coordinated gold atoms. Accordingly, their optimized structures also display some short Au···H distances, shorter than the sum of their van der Waals radii of 2.86 Å,<sup>59</sup> and the corresponding Au···H–C angles are about 120° (Fig. 2a and S7†). In comparison with the  $[\text{Au}_6]^{2+}$  nanoclusters reported by Konishi,<sup>60</sup> the Au···H–C interactions are relatively weak, which may be due to the smaller Au···H–C angles in **R**/S-**Au8c**. Taking **R-Au8c** as an example, noncovalent interaction analysis was conducted using Multiwfn 3.7 (dev) to investigate the Au···H–C interactions (Fig. S8†).<sup>61</sup> Benefiting from these intramolecular Au···H–C interactions, the metallic kernels of two **Au8c** enantiomers can be “locked in” to some extent and can facilitate the enantioselectivity. The chiral metal kernels give feedback to the surficial  $\text{o-HMBA}^-$  ligands through coordinated bonds, which boosts “chiral transfer” between BINAP and  $\text{o-HMBA}^-$ .

### ESI and CID mass spectrometry of **Au8c**

Positive-ion mode ESI-MS of **Au8c** dissolved in  $\text{CH}_3\text{OH}$  was conducted to confirm the composition and charge state (Fig. 2b). It displayed a series of peaks corresponding to multi-charge state species from +1 to +3 in the  $m/z$  range of 500 to 5000 (Fig. 2a, S9 and Table S3†). The most dominant peak **1a** at  $m/z$  1874.68 can be assigned to the molecular ion peak  $[\text{Au}_8(\text{-BINAP})_3(\text{o-HMBA})_2]^{2+}$  (calc. 1874.67). Therefore, the metallic kernels of **Au<sub>8</sub>** nanoclusters carry four free electrons and exhibit a typical prolate shape based on superatom electronic theory.<sup>62,63</sup> There is +1 charge species **1c** centered at  $m/z$  3748.34, corresponding to  $[\text{Au}_8(\text{BINAP})_3(\text{o-HMBA})_1(\text{o-MBA})_1]^{+}$  (calc. 3748.32), resulting from the deprotonation of one  $\text{o-HMBA}^-$  in species **1a**. Besides, the  $\text{o-HMBA}^-$  on the metallic surface with the uncoordinated –COOH group can provide a platform to construct dimers, trimers, and polymers with the bridge of hydrogen bonds.<sup>13,64</sup> As shown in Fig. 2b, we observed +3 charged novel species at  $m/z$  2499.56 (**1b**, calc. 2499.55) which can be identified as an **Au8c** dimer possibly bonded by intermolecular hydrogen bonds.

To prove that the above-mentioned hydrogen-bonded dimer indeed exists in solution, we trapped this species **1b** for further CID mass spectrum analysis with  $\text{N}_2$  gas in the collision energy of 0–15 eV (Fig. 2c), given that CID is a versatile technique to qualitatively probe hydrogen bonds.<sup>65</sup> Notably, even when the collision energy is 0 eV, **1b** can spontaneously dissociate into **1a** (Fig. 2c and d), revealing the breakage of very weak bond

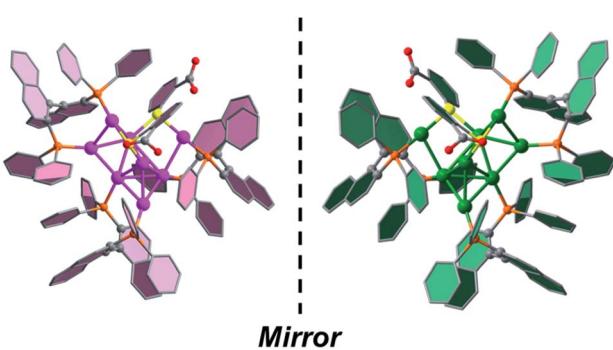


Fig. 1 Single-crystal structures of dicationic **R**-**Au8c** and **S-Au8c** nanoclusters. Color labels: pink and green, Au; yellow, S; orange, P; gray, C; red, O. All hydrogen atoms are omitted for clarity.



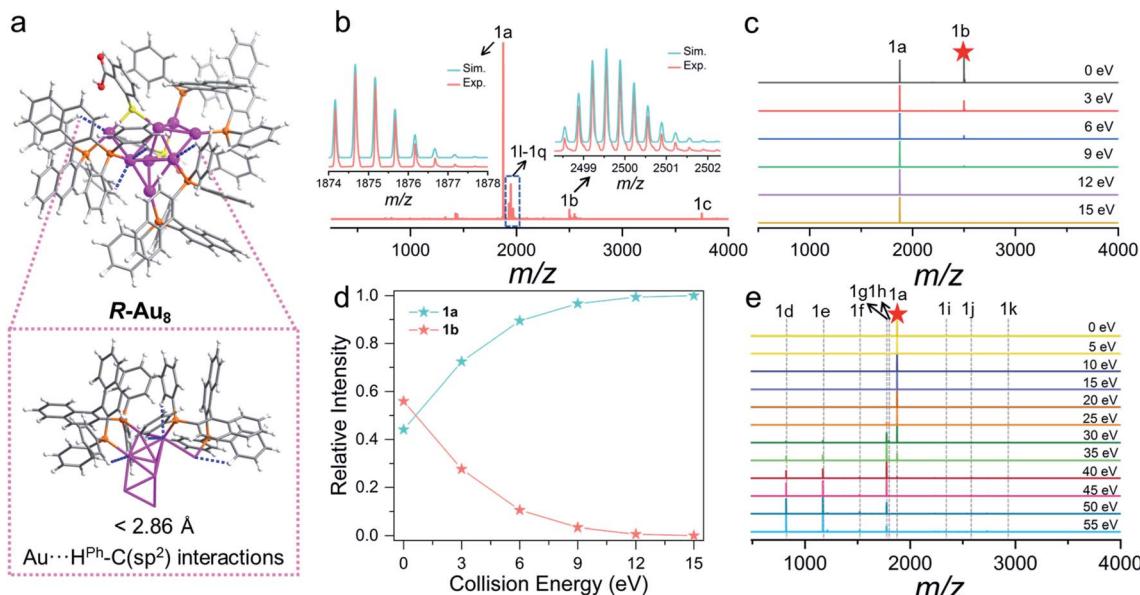


Fig. 2 (a) Optimized structure of **R-Au8c** showing  $\text{Au}\cdots\text{H}-\text{C}$  interactions. (b) Positive-ion mode ESI-MS of the **Au8c** dissolved in  $\text{CH}_3\text{OH}$ . Inset: the experimental (red trace) and simulated (cyan trace) isotopic patterns (peak **1a** and **1b**). (c) CID mass spectra of **1b** measured at the collision energy of 0–15 eV. (d) Relative intensity of the resulting **1b** fragment ions collected at the collision energy of 0–15 eV. (e) CID mass spectra of **1a** measured at the collision energy of 0–55 eV.

strength. Considering the crystallographic feature of **Au8c**, we can qualitatively infer that **1b** is a hydrogen-bonded dimer. The relative intensity of monomeric **Au8c** vs. dimeric **Au8c** is dynamically changeable and highly dependent on the collision energy. Upon increasing the collision energy from 0 eV to 15 eV, the **1b** peak progressively faded away, while the **1a** peak maintained a continuous rise to be the main species. In brief, such a CID collision-energy-dependent depolymerization of the **Au8c** dimer indicates the existence and stability of hydrogen-bonded dimer species in solution,<sup>65</sup> conducive to uncovering the probable assembly mechanism toward helical superstructures (see below).

In addition, to understand bond fission in monomeric **Au8c**, we also targeted the molecular ion peak **1a** to perform CID experiments. By plotting the relative intensity of dominant fragments as a function of collision energy (Fig. 2e), it was noticed that the relative intensity of target **1a** was almost unchanged when the collision energy was less than 15 eV and it began to decrease sharply when the collision energy continued to increase, which indicates that **1a** is quite stable in low-energy CID processes. Upon increasing the collision energy, as shown in Fig. 2e, a series of novel fragment peaks **1d**–**1k** sprang up, presenting various relative intensities. Intriguingly, all larger species **1h**–**1k** possess four valence electrons as **1a**, implying their higher stability.<sup>62</sup> Correlating the total CID mass spectra of **1a** to the relative intensity distribution of the varisized fragments (Fig. S10a†), three possible types of gas-phase dissociation patterns, such as ligand fragmentation, core fission, and ligand loss, were summarized (Fig. S10b†).<sup>66,67</sup> Each pattern competed with each other to be dominant in the collision dissociation process. Before 40 eV the propensity of **1a** is to undergo ligand fragmentation via C–H bond activation to

generate **1g** along with the loss of neutral  $\text{AuH}$ .<sup>68,69</sup> Then core fission would be the primary dissociation pathways of **1a**, displayed as the following equation:  $[\text{Au}_8(\text{BINAP})_3(o\text{-HMBA})_2]^{2+} \rightarrow [\text{Au}_n(\text{BINAP})_1(o\text{-HMBA})_m]^{+} + [\text{Au}_{8-n}(\text{BINAP})_2(o\text{-HMBA})_{2-m}]^{+}$  ( $n = 1, 2, 3$ ,  $m = n - 1$ ); the relative intensities of smaller fragments  $[\text{Au}_n(\text{BINAP})_1(o\text{-HMBA})_m]^{+}$  (**1d**, **1e**, **1f**) far exceed those of larger fragments  $[\text{Au}_{8-n}(\text{BINAP})_2(o\text{-HMBA})_{2-m}]^{+}$  (**1i**, **1j**, **1k**). In terms of core fission patterns, when  $n = 1$  or 2, the corresponding dissociation pathway may be preponderant because the relative intensity of  $[\text{Au}_n(\text{BINAP})_1(o\text{-HMBA})_m]^{+}$  is higher than that when  $n = 3$ .<sup>70</sup> In addition, **1a** can also dissociate an  $o\text{-HMBA}^-$  ligand via  $\text{Au-S}$  bond breaking, and lose a hydrogen proton to generate species **1h** ( $[\text{1a}-o\text{-HMBA}^--\text{H}^+]^{2+}$ ). More details are provided in the ESI (Fig. S10, S11 and Table S4†).

#### UV-vis spectra and electronic structure of **Au8c**

To explore the relationship between the crystal structures and optical absorption properties of **Au8c** nanoclusters, we subsequently carried out time-dependent density functional theory (TD-DFT) calculations. The UV-vis absorption spectrum of **Au8c** was measured in a dilute solution of  $\text{CH}_3\text{OH}$  at room temperature. No decomposition was observed in  $\text{CH}_3\text{OH}$  stored for at least one week (Fig. S12†), suggesting good stability of individual **Au8c** in solution. The experimental UV-vis absorption spectrum is well reproduced in the calculated one with regard to the spectral profiles (Fig. 3), which can be attributed to the stable structure of individual clusters and the suitable exchange-correlation functional. As shown in Fig. 3, the experimental absorption peak at 2.76 eV (449 nm) in the low-energy region corresponds to an adjacent calculated band, which is mainly composed of two calculated excitations at 2.78 eV (446 nm) and 2.60 eV (477 nm), with strong oscillator strength values

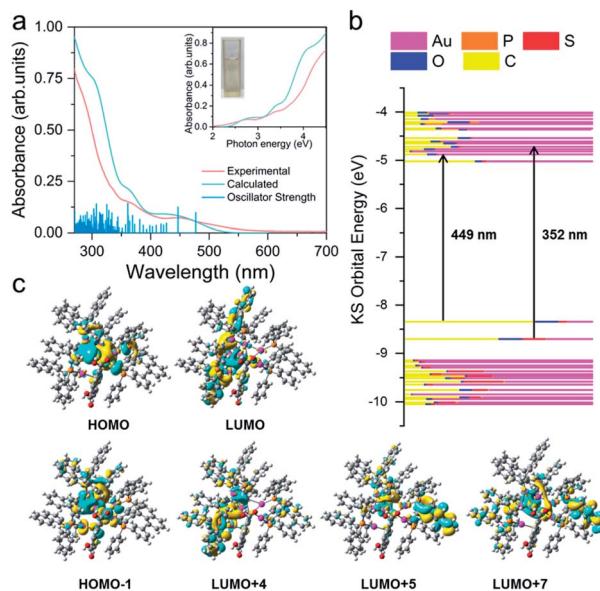


Fig. 3 (a) Experimental (pink line) and theoretical (cyan line) UV-vis absorption spectra of Au8c. Inset: experimental (pink line) and theoretical (cyan line) photon-energy plots of Au8c. (b) Kohn-Sham molecular orbital energy level diagram and the associated populations of atomic orbitals in each KS molecular orbital for R-Au8c. (c) The calculated HOMO, HOMO-1, LUMO, LUMO+4, LUMO+5 and LUMO+7 orbitals of R-Au8c.

of 0.0416 and 0.0326, respectively. Molecular orbital analysis suggested that the most likely transitions are the electronic transitions between the HOMO, which is mainly localized on the 5d and 6s atomic orbitals of Au atoms, and several energetically close lying LUMOs (LUMO+1/LUMO+4/LUMO+5/LUMO+7), involved in the metal-to-ligand charge transfer (MLCT) and metal-to-metal charge transfer (MMCT) (6s/5d → 5p). In the high-energy region, there are two experimental bands centered at 3.52 eV (352 nm) and 4.22 eV (294 nm), corresponding to the calculated absorption bands centered at 3.43 eV (361 nm) and 4.03 eV (307 nm), which have oscillator strength values of 0.0475 and 0.0467, respectively. The absorption band of 3.43 eV primarily originates from HOMO-1 → LUMO+4/LUMO+6 and HOMO → LUMO+9 transitions, which mainly comprise MLCT and a small amount of ligand-to-ligand charge transfer (LLCT) from the p orbitals of S atoms in *o*-HMBA<sup>-</sup> to the  $\pi^*$  orbitals of BINAP ligands. As for the calculated absorption band at 4.03 eV (307 nm), the transitions are more complicated because multiple orbitals are involved. More calculated details of Au8c are listed in Fig. S13 and Tables S5–S7.<sup>†</sup> Noteworthily, the diverse electronic transitions existing in Au8c provide essential information for us to understand the electronic structure of Au8c as well as for the subsequent investigations on the origin of chirality.

### Chirality of R/S-Au8c

Next, we measured the CD spectra of R-Au8c and S-Au8c in a diluted CH<sub>3</sub>OH solution. As shown in Fig. 4, the calculated and experimental CD spectra of Au8c enantiomers both show perfect mirror image chiroptical responses. Taking R-Au8c as

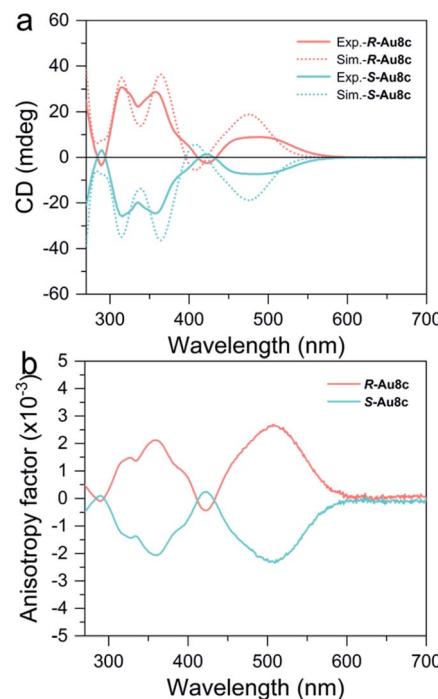


Fig. 4 (a) The CD spectra of R-Au8c (pink trace) and S-Au8c (cyan trace) dissolved in CH<sub>3</sub>OH. (b) Corresponding anisotropy factor of R-Au8c (pink trace) and S-Au8c (cyan trace). Dot lines: simulated CD spectra.

a representative, the experimental CD spectrum has two sharp signal peaks at 315 and 358 nm, and a broad signal peak approximately at 486 nm, which are in good agreement with the calculated ones. The experimental broad CD peak of R-Au8c at the low energy gives the maximum anisotropy factor  $g_{abs}$  value reaching  $2.5 \times 10^{-3}$ , which is higher than that in our previous report on BINAP-protected Au<sub>19</sub> and Au<sub>11</sub> nanoclusters,<sup>71</sup> whose chiroptical activities originate from the distorted core and chiral ligands. Such a CD peak corresponds to the experimental absorption peak at 449 nm, and it is related to the MLCT transition as well as the MMCT transition of the inherently chiral metal core. This suggests that the inherently chiral Au<sub>8</sub> kernel plays a crucial role in enhancing the  $g_{abs}$  values. Moreover, two CD signals at 315 and 358 nm in the high-energy band also can match with the absorption peaks at 294 nm and 352 nm, respectively, which mainly arise from the MLCT transition from the Au<sub>8</sub> kernel to BINAP/*o*-HMBA<sup>-</sup> ligands together with small LLCT transition involving chiral BINAP to *o*-HMBA<sup>-</sup>. Furthermore, the continuous chirality measure (CCM) values of the metallic kernel and the metallic kernel coordinated by S and P atoms of R-Au8c and S-Au8c were calculated to account for the chiral origins.<sup>72,73</sup> The CCM value of the Au<sub>8</sub> kernels in R-Au8c and S-Au8c is 12.60, indicating their inherently chiral metal kernel. When the surface of Au<sub>8</sub> kernels was ligated with S and P atoms from ligands, the CCM values show only a slight increase (Table S8<sup>†</sup>). These results reveal that the chirality of individual Au8c is not only derived from the ligation of chiral BINAP ligands, but also from the intrinsic chiral metal kernel.



## Crystalline state helical self-assembly of chiral Au8c

In nature, typical secondary structures exist in peptides or proteins through helical ordered arrangements of simple primary structures.<sup>74</sup> Notably, the inserted  $(\text{HCO}_3^-)_2$  dimers offer a good bridge to connect adjacent  $\text{Au}_8$  nanoclusters so that individual **Au8c** enantiomers can transfer their chirality and form helical superstructures during the self-assembly process. Along the *b* axis, neighbouring **Au8c** nanoclusters are locked together compactly *via* rich  $\text{O}-\text{H}\cdots\text{O}$  hydrogen bonds supported by the  $(\text{HCO}_3^-)_2$  dimers and the  $-\text{COOH}$  group of *o*-HMBA<sup>-</sup> ligands on the surface of **Au8c**, namely  $\text{O}-\text{H}\cdots\text{O}$  ( $-\text{COOH} \rightarrow (\text{HCO}_3^-)_2$  and  $(\text{HCO}_3^-)_2 \rightarrow -\text{COOH}$ ) (Fig. 5a). The average  $\text{O}-\text{H}\cdots\text{O}$  distance in **R-Au8c** is 2.72 Å (range: 2.65–2.79 Å) with an average  $\text{O}-\text{H}\cdots\text{O}$  bond angle of 146.88° (range: 126.6–159.8°) (Table S9†). Driven by these  $\text{O}-\text{H}\cdots\text{O}$  hydrogen bonds, the **Au8c** enantiomers imitate the behaviour of biomolecules to assemble into helical secondary structures along the  $2_1$  screw axis (Fig. 5b). **R-Au8c** and **S-Au8c** exhibit an identical stacking modality, but generate left-handed (*M*-single) and right-handed (*P*-single) helical superstructures, respectively (Fig. 5a). To the best of our knowledge, such helical architectures driven by hydrogen bonded interactions located between gold nanoclusters and inorganic ions are still very rare.

To understand the crucial role exerted by inserted  $(\text{HCO}_3^-)_2$  dimers in constructing helical superstructures, a comparative analysis of **R-Au8d** without  $\text{HCO}_3^-$  was performed. In the same direction (*b* axis), **R-Au8d** also exhibits a helical arrangement running along the  $2_1$  screw axis (Fig. S14†). Varying from **R-Au8c**, due to the deprotonation of *o*-H<sub>2</sub>MBA ligands and the absence of  $\text{HCO}_3^-$  anions, the helical superstructure of **R-Au8d** is stabilized by  $\text{C}-\text{H}\cdots\text{O}$  (average: 2.588 Å) and  $\text{C}-\text{H}\cdots\pi$  (2.808 Å) interactions (Fig. S14†). Additionally, compared to **R-Au8c**, the stability of the **R-Au8d** helix should be weaker because the  $\text{O}-\text{H}\cdots\text{O}$  hydrogen bond interactions are generally stronger than

$\text{C}-\text{H}\cdots\text{O}$  and  $\text{C}-\text{H}\cdots\pi$  interactions.<sup>75</sup> The above results indicate that the inserted  $(\text{HCO}_3^-)_2$  dimers are indispensable for the formation of stable helical superstructures.

Moreover, it should also be noted that even a slight difference in positional isomeric *o*-H<sub>2</sub>MBA and *m*-H<sub>2</sub>MBA ligands can make structures and packing patterns different for  $\text{Au}_8$  nanoclusters (**R-Au8d** *versus* **R-Au8e**, Fig. S15†). In contrast to the case of **R-Au8d**, the weaker  $\text{C}-\text{H}\cdots\text{O}$  interactions (average: 2.663 Å) are the dominant driving force for the formation of the helical superstructure of **R-Au8e**. This also implies a special role of the inserted  $(\text{HCO}_3^-)_2$  dimers in the hydrogen-bonded helices of **Au8c**.

Furthermore, the chirality of crystalline-state **Au8c** was also investigated by CD spectroscopy. To eliminate the linear dichroism (LD) effect caused by crystal birefringent, the crystals of **Au8c** were adequately ground and mixed with KBr, and the mixtures were further pressed into tablets for testing. Compared to their CD spectra in diluted solution with three CD peaks, the CD spectra of crystalline-state **Au8c** are simpler in the same range. As shown in Fig. 5c, the *P*-helical **R-Au8c** displays two dominant Cotton effect bands at 368 nm (positive) and 517 nm (positive), while the *M*-helical **S-Au8c** affords a completely inverse CD spectrum. The discrepancy in the CD spectra of the solution and crystalline state can be related to the distinct intermolecular interactions. For **Au8c**, the hydrogen bonds and electrostatic interactions between individual  $\text{Au}_8$  nanoclusters and  $(\text{HCO}_3^-)_2$  dimers should play a key role in the crystalline state.

As indicated by ESI-MS, the most dominant species in  $\text{CH}_3\text{OH}$  is monomeric **1a** ( $[\text{Au}_8(\text{BINAP})_3(\text{o-HMBA})_2]^{2+}$ ), accompanied by  $\text{Au}_8$  dimers with an extremely weak abundance, suggesting that the helical superstructures didn't exist in  $\text{CH}_3\text{OH}$  ( $8.0 \times 10^{-6}$  to  $3.0 \times 10^{-4}$  M), which is also evidenced by concentration-dependent IR and CD spectra of **R-Au8c** in  $\text{CH}_3\text{OH}$  (Fig. S16 and S17†).

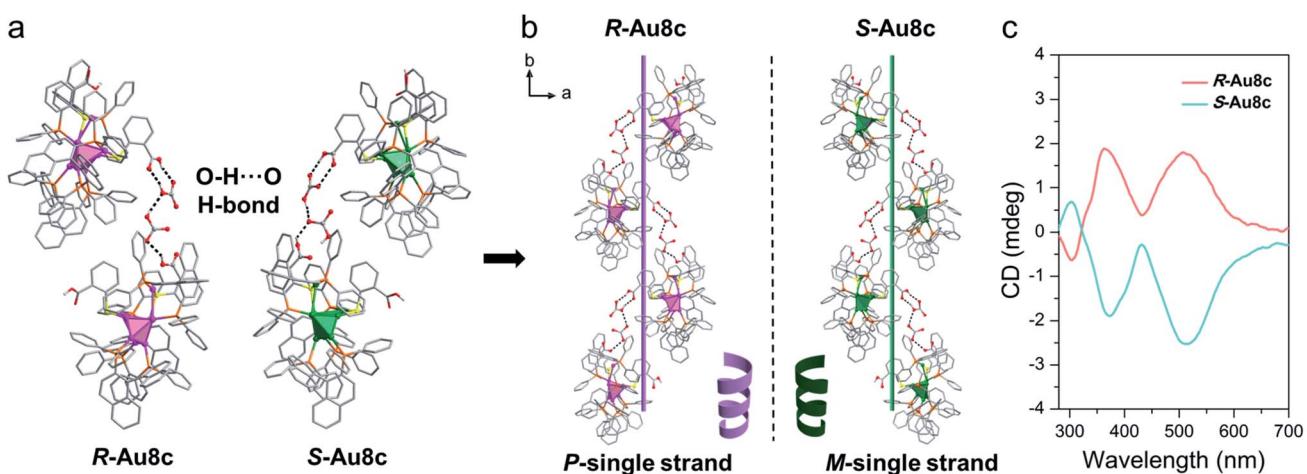


Fig. 5 (a) The  $\text{O}-\text{H}\cdots\text{O}$  hydrogen bonds existing between two adjacent **R-Au8c** and **S-Au8c** involving the  $(\text{HCO}_3^-)_2$  dimer. (b) Perspective diagrams of supramolecular chirality for **R-Au8c** and **S-Au8c** indicating the asymmetric packing into *P*- and *M*-single strands, respectively. (c) CD spectra of **R-Au8c** (pink line) and **S-Au8c** (cyan line) in the crystalline state under ambient conditions. Color labels: pink and green, Au; orange, P; yellow, S; gray, C; red, O; white, H.



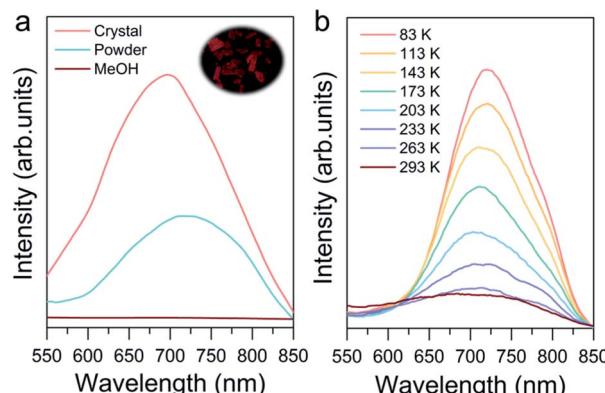


Fig. 6 (a) Emission spectra of crystals (pink line), powder (cyan line) and CH<sub>3</sub>OH solution (brown line) of Au8c. Inset: photograph of Au8c crystals under green light excitation. (b) Temperature-dependent emission spectra of Au8c in the crystalline state ( $\lambda_{\text{ex}} = 365$  nm).

### Photoluminescence properties of Au8c

Photoluminescence properties were also investigated to explore their potential applications such as in imaging, sensing and cell labeling.<sup>76–78</sup> As shown in Fig. 6a, the crystalline state **Au8c** exhibited emission at 698 nm at room temperature under excitation at 365 nm. In contrast, the CH<sub>3</sub>OH solution of **Au8c** is nonemissive because of the chaos of Au<sub>8</sub> nanoclusters in solution, which prevents the formation of ordered intermolecular interactions, suppressing the radiation decay route. For comparison, a powder sample was prepared by rotary evaporation of CH<sub>3</sub>OH solution of **Au8c**, and its luminescence intensity was weaker than that in the crystalline state along with a red-shift of 20 nm. This can be rationalized by the limited and chaotic intermolecular interactions in powder where a looser packing of the nanoclusters is formed.<sup>79</sup> All of these results refer to a CIEE phenomenon in **Au8c**.<sup>48,79</sup> The compact intermolecular hydrogen bonds as well as the electrostatic interactions distributed in **Au8c** might restrict the intra-molecular rotations and vibrations, which is attributed to the boosted luminescence. In summary, the special arrangement of **Au8c** nanoclusters in the crystal lattice plays the key role in determining its photoluminescence behavior. Additionally, we also observed temperature-dependent emissive behavior of **Au8c** in the crystalline state (Fig. 6b). Upon decreasing the temperature from 293 to 83 K, the intensity of the emission peak gradually increased along with a red-shift from 698 to 718 nm. A similar phenomenon has also been observed in other coinage metal clusters.<sup>80</sup> We anticipate that **Au8c** with the near-infrared (NIR) luminescence feature can serve as a promising material for practical applications.

### Conclusions

In summary, we successfully synthesized a pair of atom-precise gold nanocluster enantiomers, **R-Au8c** and **S-Au8c**, through an elaborate selection of the hydrophobic chiral BINAP and hydrophilic achiral *o*-H<sub>2</sub>MBA ligands as protective ligands.

Based on the single crystal structures, CD spectra in solution and DFT analysis, the chirality of individual **Au8c** nanoclusters is mainly contributed by the inherently chiral metallic kernel and BINAP ligands. Driven by intermolecular hydrogen bonds between (HCO<sub>3</sub>)<sub>2</sub> dimers and individual Au<sub>8</sub> nanoclusters, **R-Au8c** and **S-Au8c** demonstrate *P*- and *M*-single stranded helical superstructures in the crystalline state, respectively. This work demonstrates a structural model for in-depth understanding of the self-assembly mechanisms of cluster-based helical superstructures and provides an effectively designed approach to construct atom-precise helical supramolecular materials. Future work will focus on exploring the potential applications of these helical superstructures and constructing more diverse helical superstructures.

### Data availability

All experimental and computational data associated with this article have been included in the main text and ESI.†

### Author contributions

D. S. conceived the manuscript; W. D. S. and K. S. performed the experiments; W. D. S., K. S., C. Z., Z. W., S. S. Z., J. M. D., L. F., Z. Y. G. and D. S. analyzed data, prepared figures and provided conceptual contributions; W. D. S., K. S., C. H. T. and D. S. wrote the manuscript with contributions from all co-authors.

### Conflicts of interest

The authors declare no competing financial interest.

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