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Water splitting by $\text{MnFe}_2\text{O}_4/\text{Na}_2\text{CO}_3$ reversible redox reactions†

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Future energy systems must call upon clean and renewable sources capable of reducing associated CO_2 emissions. The present research opens new perspectives for renewable energy-based hydrogen production by water splitting using metal oxide oxidation/reduction reactants. An earlier multicriteria assessment defined top priorities, with $\text{MnFe}_2\text{O}_4/\text{Na}_2\text{CO}_3/\text{H}_2\text{O}$ and $\text{Mn}_3\text{O}_4/\text{MnO}/\text{NaMnO}_2/\text{H}_2\text{O}$ multistep redox cycles having the highest potential. The latter redox system was previously assessed and proven difficult to be conducted. The former redox system was hence experimentally investigated in the present research at the 0.5 to 250 g scale in isothermal thermogravimetry, an electrically heated furnace, and a concentrated solar reactor. Over 30 successive oxidation/reduction cycles were assessed, and the H_2 production efficiencies exceeded 98 % for the coprecipitated reactant after these multiple cycles. Tentative economics using a coprecipitated reactant revealed that 120 cycles are needed to achieve a 1 € per kg H_2 cost. Improving the cheaper ball-milled reactant could reduce costs by approximately 30 %. The initial results confirm that future research is important.

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1. Introduction

The future annual worldwide need for H_2 was estimated by Liu *et al.*:¹ reducing global CO_2 emissions requires that at least 10% of hydrocarbons (currently 14 billion tons of oil equivalent²) be substituted by alternative renewable energy resources, including “green” H_2 . If only H_2 is considered as a substitute energy carrier, approximately 1000 Mtons of nonfossil-based H_2 are needed. Current H_2 production is more than 10 times below the required 10% replacement target,³ but electrolysis using renewable energy is promoted. Neither hydropower³ nor nuclear energy are sufficient to meet this target.⁴ The nuclear energy production was only 25 exajoule (even less than 39 exajoule per year of hydropower). The negative public attitude towards nuclear energy and the limited reserves of uranium will probably not foster increased nuclear power.

Currently, green electrolytic H_2 can be produced using mostly wind or photovoltaic electricity. Its production

requires an average electricity consumption of approximately $50 \text{ kW h kg}^{-1} \text{ H}_2$ at an overestimated efficiency of 80%.⁵ The 2019 global renewable energy production, without hydro-energy, was estimated at 2800 TW h. If these sources were devoted to solely produce H_2 , only 56 Mtons of electrolysis H_2 could be generated, and this at a 2- to 4-fold production cost of the traditional petrochemical methods. Renewable technologies are therefore unable to meet the 1000 Mtons H_2 goal.

It is therefore expected that fossil hydrocarbons will remain the important H_2 sources in the near future, despite their environmental impact through CO_2 emissions. Steam methane reforming yields approximately 10 kg of CO_2 per kg H_2 , and is hence classified as “grey hydrogen” production method.³ The application of carbon capture and storage could make this technology “cleaner”, but involving significant costs to nearly double the price of the produced H_2 .⁶

Within the “green” H_2 production methods,³ thermochemical water splitting by redox systems⁵ has a considerable potential, and is the subject of the present research. Alternative methods were, however, also investigated.^{7,8}

1.1 H_2 from multicycle thermal redox water splitting

Thermochemical redox water splitting reactions are candidates for H_2 production. Water is abundant, inexpensive, and its decomposition or subsequent combustion is free from CO_2 emissions. Thermochemically splitting water into H_2 and O_2 involves different reaction pathways, but the overall endothermic reaction ($\Delta H = 286 \text{ kJ mol}^{-1}$) is $\text{H}_2\text{O} \rightarrow \text{H}_2 + \frac{1}{2}\text{O}_2$.

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These thermochemical reactions to produce H_2 offer major advantages in comparison with common alkaline water electrolysis, which has a low efficiency of approximately 20% (30% for electricity and 65% for electrolysis). The high-temperature thermochemical H_2 production efficiency is much higher, provided cheap renewable or waste high-temperature heat is available.

These water splitting systems have been proposed since 1964. The initial vanadium–chlorine cycle was, however, abandoned for its low efficiency, high cost, and the formation of toxic and hazardous products. The cycles were however further developed, and approximately 25 thermochemical cycles are currently proposed. Deng *et al.* examined and ranked these proposed redox reactions by multiple criteria assessment, including quantified parameters of thermal, chemical, environmental and economic nature.⁵

Very high temperature reactions ($\gg 1000^\circ\text{C}$) of metal–metal oxides/hydroxides, doped ceria, or perovskites were not considered since such high temperatures would necessitate the application of high-cost alloys to construct redox reactors. Selected redox systems should operate at temperatures that would limit reactor wall temperatures below 1000°C , thus operating the redox material bed at maximum 800°C .⁵ Based on this important target, 4 out of 24 redox reactions were finally selected, and included $\text{MnFe}_2\text{O}_4/\text{Na}_2\text{CO}_3$, $\text{Mn}_3\text{O}_4/\text{MnO}/\text{NaMnO}_2$, $\text{U}_3\text{O}_8/\text{UO}_2\text{CO}_3$ and $\text{ZnO}/\text{Fe}_3\text{O}_4/\text{ZnFe}_2\text{O}_4$. The U_3O_8 cycles were discarded due to nuclear hazards. The ZnO cycles scored significantly lower than both remaining redox systems, and were not further investigated. The $\text{Mn}_3\text{O}_4/\text{MnO}/\text{NaMnO}_2$ four-step redox process was assessed, but required an operation temperature of approximately 800°C and suffered from poor

reversibility of the redox cycles.¹ The $\text{MnFe}_2\text{O}_4/\text{Na}_2\text{CO}_3$ cycle remained the selected system under scrutiny.

1.2 Solar heat-induced redox systems

The use of solar energy in the redox cycles for H_2 production was previously advocated.^{9,10} Metal oxide redox pairs are considered the simplest, and the most environmentally friendly. Solar heat supplies the sensible and reaction heats of the oxidation and reduction cycles. The ferrite system was shown to split water at low temperatures in the oxidation reaction and reduce CO_2 in the reduction reaction.^{11–13} Solar-induced two-step water splitting by (Ni, Mn) ferrite and $\text{ZnO}/\text{MnFe}_2\text{O}_4$ was achieved at approximately 1000 K .¹⁴ The solar-based ZrO_2 -supported Co(II)-ferrite cycle was finally tested but the required temperatures of 1000°C for the oxidation and 1400°C for the reduction, and the attrition of the reactant during the reduction were prohibitive for its further application. Carbon-bearing systems failed in repeated cycles.^{12,13}

1.3 Literature findings on MnFe_2O_4 and its applications

Spinel ferrites (MnFe_2O_4) are a significant class of magnetic materials, with excellent electrical properties. Their development is therefore mainly linked with electronic uses, for example shifters, frequency transformers, PCs, TV, cell phones, adsorbents, among other applications. Their use in water splitting was launched during the past decade.

The manufacturing methods are diverse, with main objectives to produce MnFe_2O_4 particles of small size, high crystallinity, and large specific surface area. The comparison of the manufacturing methods and the relevant physicochemical properties is given in Table 1. The manufacturing methods used

Table 1 Comparison of physicochemical properties of MnFe_2O_4 particles synthesized *via* different methods

Synthesis method	Reaction T ($^\circ\text{C}$)	Time (h)	Particle size (nm)	Surface area ($\text{m}^2\text{ g}^{-1}$)	Ref.
Polyol	210	3	7	165.39	15
Co-precipitation	70	2	20–80	84.5	16
	95	2	30	—	17
	50	3	24	—	18
	70	1	36	—	19
	75–80	6	14	0.293 (ref. 20)	21
	80	6	80	—	22
Sol-gel	70	2	45	—	19
Ball milled	900	1	6.78–8.06	—	23
Thermal decomposition	295	1	7	—	24
	270	1.5	18.9	—	25
Solution combustion	300	—	30–35	33	26
Hydrothermal	200	12	30–50	—	27
	220	10	14.5	—	28
	180	28	280	32.19	29
	180	16	16	—	19
	200	24	8.6	—	30
Solvothermal	180	12	60	70	31
	500–600	24	12–22	—	32
	200	8	250–260	—	33
Co-precipitation	80	2	40	132	This work
Ball milled	20	1	100	4.92	This work



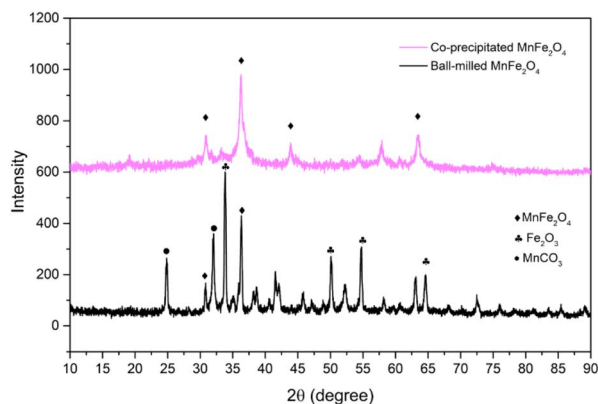


Fig. 1 Diffraction patterns of the selected reactants.

in the present research, *i.e.* by simple ball-milling and by co-precipitation, are included in the table.

Whereas ball-milled reactant appears less performance due to its coarse particle size and low surface area, coprecipitated reactant seems to meet the main property targets. Our coprecipitation differs from other cited synthesis methods through its use of metal chlorides as precursors, rather than the previously favoured metal nitrates or sulphates.

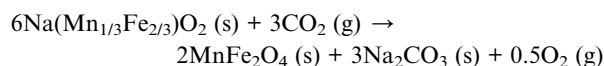
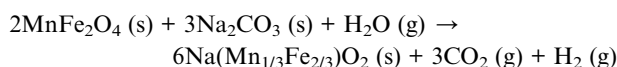
The crystalline of both ball-milled and coprecipitated MnFe_2O_4 , is determined from XRD analysis. The diffraction patterns of both reactants are shown in Fig. 1.

In both cases, diffraction peaks were matched with the respective references of the ICDD cards. The main peaks correspond to the space group $Fd\bar{3}m$ (spinel ferrite). The crystallite size for each phase was determined from the Scherrer's equation,³⁴ accounting for the most intense XRD-peak. The crystal size, δ , is determined as:

$$\delta = \frac{k\lambda}{I(2\theta)\cos\theta}$$

with, k , as a correction factor, commonly 0.9; λ as the wavelength of X-ray diffraction of the α -Cu electrode (0.154 nm); and θ is the Bragg angle (degree). Therefore, the average crystallite size of ball-milled MnFe_2O_4 is determined to be about 15.5 nm, and the co-precipitated MnFe_2O_4 is about 10.4 nm. It can be noted that the crystallite sizes are very similar, despite the different synthesis method.

The fairly novel application in water splitting relies on the thermochemical cycle of $\text{MnFe}_2\text{O}_4/\text{Na}_2\text{CO}_3$, as described by the following reactions:¹



The optimal operation temperature of both reactions is approximately 700 to 750 °C.³⁵ H_2 production at lower

temperatures shows slower kinetics, and complete regeneration cannot be fully accomplished.^{36,37} The reduction step is problematic and less effective, although adding Fe_2O_3 could help to overcome the problem.³⁸ The reduction step seems to pose a main problem for all the redox pairs that were reported,^{35,39} and the reaction kinetics are not fully defined. The problems in the oxygen-releasing step are amplified along with the increasing experimental scale.

Although literature is still scarce, earlier research was reported by Murmura *et al.*,³⁹ and by Varsano *et al.*³⁵ Both Murmura *et al.*³⁹ and Varsano *et al.*³⁵ used about 40 mg $\text{MnFe}_2\text{O}_4/\text{Na}_2\text{CO}_3$ at the lab-scale for temperatures between 700 and 800 °C, or 600 and 800 °C, respectively. The H_2 yield obtained after 1 h of oxidation varied from 81% (700 °C) to 86% (800 °C).

Varsano *et al.*⁴⁰ repeated the tests in a 1 kW concentrated solar facility at 750 and 800 °C and obtained H_2 yields of approximately 20 to 37% at 750 °C, and ~72% at 800 °C. The lower H_2 efficiency in the solar-driven reactor was attributed to a nonuniform temperature within the solar reactor, and to a too coarse particle size (0.5–2 mm) used in the solar reactor.⁴¹ Such coarse particles have a very low active surface area. In general, the previous studies paid insufficient attention to the individual steps (oxidation and reduction), to the reaction kinetics, and to the long-duration and multicycle operation.

1.4 Objectives of the research

The art-to-date studies of redox systems were mostly limited to single steps of the cycles, with experiments performed on the milligram scale, and without considering long-duration cyclability and process economics. The present research will for the first time assess water splitting on a small and a larger scale, by thermogravimetric analysis (TGA), in an isothermal electric furnace, and in a concentrated solar rig. Realistic operating conditions and equipment scale will be accounted for. The study is limited to the priority-selected $\text{MnFe}_2\text{O}_4/\text{Na}_2\text{CO}_3$ cycles.¹ It is expected that the results will provide better insight into thermochemical H_2 production in equipment of different scales. Special attention will be given to the results obtained in multiple cycles, hence providing a better view of process economics, and scaling up.

2. Materials and methods

2.1 Synthesis and main properties of redox reactants

The priority redox system under scrutiny involves a spinel MnFe_2O_4 and Na_2CO_3 according to the reaction of Section 1.3. The preparation of the $\text{MnFe}_2\text{O}_4/\text{Na}_2\text{CO}_3$ mixture involved either (i) a ball-mill method involving mostly MnCO_3 and Fe_2O_3 , or (ii) a coprecipitation method involving $\text{MnCl}_2 \cdot 4\text{H}_2\text{O}$ (manganese(II) chloride tetrahydrate) and anhydrous FeCl_3 (iron(III) chloride, coprecipitated with sodium hydroxide (NaOH)). Na_2CO_3 (>99.9% purity) was purchased from Luchi Co., Ltd. Manganese(II) carbonate (>99.9% pure), iron(II) oxide (>96% pure), MnCl_2 (>99% pure) and FeCl_3 (>98% pure), were acquired from Sigma-Aldrich Chemie GmbH.

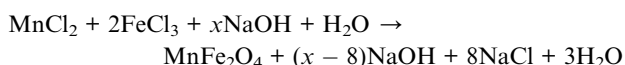


Table 2 Average volume diameter (d_v) and surface/volume diameter (d_{sv}) of the feedstock particles

Chemicals	d_v (μm)	d_{sv} (μm)	σ (μm)
MnCO_3	10.9	9.2	5
Na_2CO_3	394.5	331.3	36.5
MnCl_2	6.4	5.9	0.5
Fe_2O_3	5.6	5.3	0.3
Olivine (100–150 mesh)	167.9	142.7	18.9

In the ball-mill preparation,⁴² analytical grade MnCO_3 and Fe_2O_3 were used without further treatment. They were mixed for 40 minutes at a molar ratio of 3 : 2 in a 1000 rpm Simoloyer CM01 mixer/mill at ambient conditions and after the addition of ethanol (96%). 5 mm diameter stainless steel balls were used at a weight ratio of 6 with respect to the reactant's mix. Ethanol was evaporated at 378 K. The dried reactant was calcined in N_2 atmosphere at 973 K for 1 hour, and re-milled in a Retsch mill. The calcination under N_2 is needed to activate the reactants before their first use.

For the coprecipitated MnFe_2O_4 , an aqueous solution of 0.5 mol MnCl_2 and 1 mol FeCl_3 were mixed at 60 °C with continuous stirring at 250 rpm. Subsequently, a 0.64 mol NaOH solution was added. The solution was maintained at 80 °C for 1 h. The precipitates were centrifuged and washed 5 times with distilled water at 80 °C (to remove excess NaOH and formed NaCl). After drying at 105 °C and calcination at 450 °C during 2 h, the powder was milled in a Retsch mill into fine particles (<50 μm). The chemical reaction is presented as



For the experiments in the vertical and solar furnaces, inert olivine was added to form a porous fixed bed, or to be able to operate the solar reactor in an isothermal fluidized mode by improving the heat transfer from the reactor wall to the fluidized bed of reactants. Malvern laser-diffraction and confirming SEM-imaging were used to determine the particle size

distributions, which were mostly Gaussian with a narrow size distribution. This is illustrated in Table 2 for the feedstock particles. The near-spherical olivine particles (Mg, Fe-silicates) had a Waddell sphericity factor ψ of 0.8–0.9.⁴¹ The sphericity of all reactants was considered close to 0.84. These particles were further milled and processed into smaller particles. The pulverized mixtures had an ultrafine particle size, well below 2 μm , with smaller particles of approximately 150 nm.

Particle size was measured by diffractometry, and the BET surface area ($\text{m}^2 \text{g}^{-1}$; Brunauer–Emmett–Teller) was determined in a Micromeritics instrument by low-temperature (−196 °C) nitrogen adsorption. SEM images are shown in Fig. 2 for the prepared MnFe_2O_4 compound reactants (ball-milled and coprecipitated). The ball-milled reactant particles have an agglomerated particle diameter of $\sim 60 \mu\text{m}$ (Fig. 2a) and are composed of smaller ($\sim 0.1 \mu\text{m}$) grains. Higher magnifications of coprecipitated reactant (Fig. 2b) show that smaller particles of 50 to 150 nm size are obtained. The coprecipitated reactants have a higher specific surface area ($132 \text{ m}^2 \text{g}^{-1}$) than the ball-milled reactants ($4.9 \text{ m}^2 \text{g}^{-1}$).

2.2 Experimental setups and reactants

Three experimental setups were used, an electrically-heated vertical furnace for initial guidance tests, a TGA for long-duration experiments in multiple cycles, and a pilot-scale concentrated solar rig with ongoing experiments in the solar high season. Each of the experimental setups comprises the same elements.

N_2 (carrier gas) and CO_2 feeds are set by mass flow meters. H_2O is added by a syringe pump. The $\text{N}_2/\text{H}_2\text{O}$ (oxidation cycle) or CO_2 (reduction cycle) flows are preheated before being added into the water splitting reactor. The reactors are either vertical furnaces or TGA cells. After the reaction, the exhaust gas is cooled and dehumidified before being sent to a GC-MS for component monitoring. CO_2 was removed from the gas by absorption in a $1.5 \text{ g L}^{-1} \text{Ca(OH)}_2$ solution. The reactors contained appropriate quantities of reactant particles. A full description of the setups and the applied experimental procedures are given in ESI-3.[†] Olivine was sometimes added to increase the porosity and flowability of the reactants. The ball-milled reactant was prepared by milling 7.05 g ferrite and

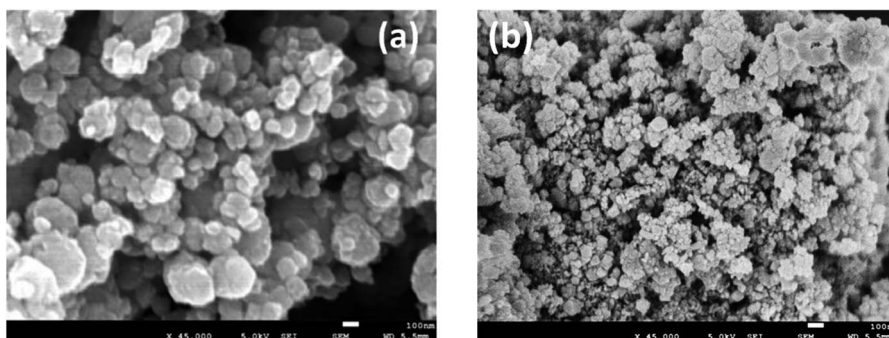


Fig. 2 SEM images of MnFe_2O_4 , as-synthesized at 700 °C (a) ball-milled; (b) coprecipitated.



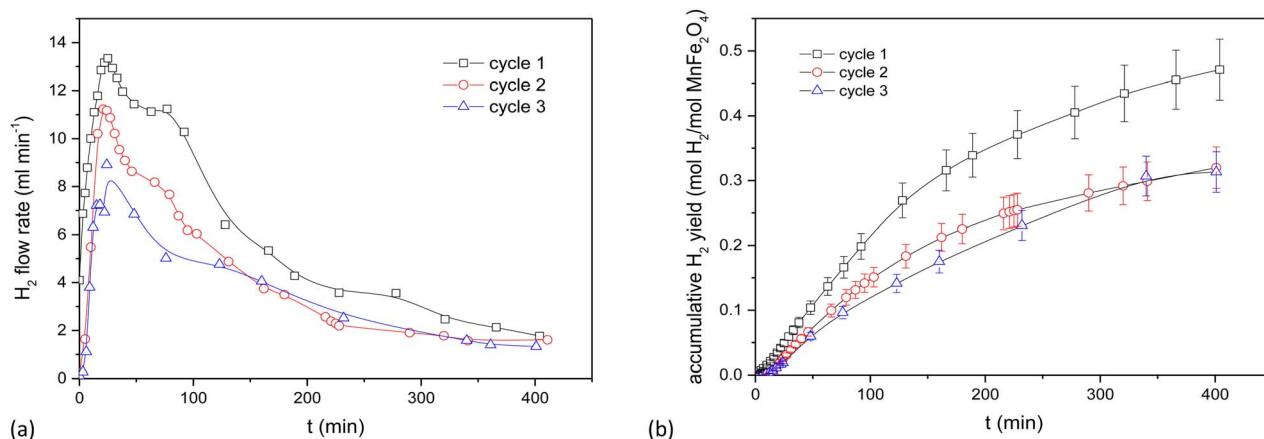


Fig. 3 Results of the WS reaction cycle (700 °C) for co-precipitated MnFe₂O₄ at different oxidation time, but with a fixed reduction cycle of 3 h. (a) WS cycles of MnFe₂O₄ (coprecipitation). (b) Cumulative results for the WS cycles of MnFe₂O₄ (coprecipitation).

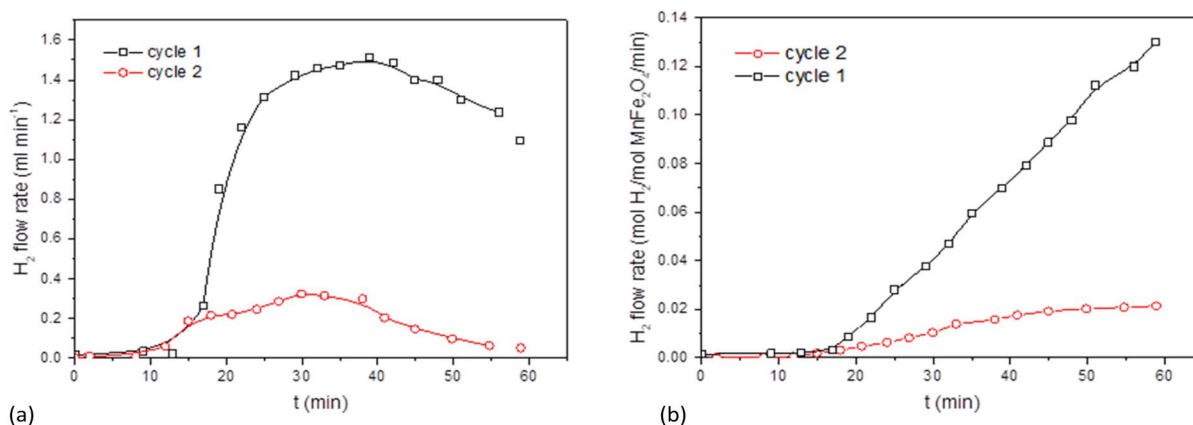


Fig. 4 Results for ball-milled MnFe₂O₄ at different oxidation time, but with a fixed reduction cycle of 3 h. (a) WS cycles of MnFe₂O₄ (ball milling). (b) Cumulative results for the WS cycles of MnFe₂O₄ (ball milling).

4.87 g Na₂CO₃. For coprecipitation reactant, 53.94 g of MnFe₂O₄ was mixed with 37.18 g of Na₂CO₃. The bed heights in the electrically heated furnace were 15 cm, against 25 cm in the solar reactor.

The GC-MS continuously monitored and recorded the H₂ concentration in the oxidation step, and the O₂ concentration in the reduction step.

3. Results and discussion

3.1 The MnFe₂O₄ system in the electrically heated furnace

The use of the electrically heated setup is tedious and requires frequent dismantling to obtain reactant samples. Its results however provide data for a relatively deep bed and facilitate a kinetic analysis at a fair sample scale.

The behaviour of the activated MnFe₂O₄ reactant was assessed for 3 subsequent oxidation–reduction steps each at 700 °C. The reverse step used pure CO₂ for 3 h. Time-dependent H₂ production values are illustrated in Fig. 3 for coprecipitated MnFe₂O₄, and in Fig. 4 for ball-milled MnFe₂O₄. The results were cumulated and expressed in mol H₂ per mol MnFe₂O₄. The

coprecipitated reactant significantly performs better than the reactant produced by ball milling.

Since it was seen that the ball-milled reactant had a significantly lower H₂-yield than its coprecipitated alternative, these tests were terminated after 60 min. It was moreover evident that reduction times longer than 3 h were needed. According to Chen *et al.*,⁴³ the oxygen release (step 2 of the reaction cycle) between layered Na(Mn_{1/3}Fe_{2/3})O₂ and CO₂ is fairly slow and requires >3 h to be completed.^{44,45} This was further investigated by TGA, where it was demonstrated that the reduction step preferably requires 4.5 to 6 h. Stoichiometrically, the yield should be 0.5 mol H₂ per mol MnFe₂O₄, which is only closely achieved for the coprecipitated reactant. Regrinding of the reactants between the cycles slightly increased the H₂ yield. It was hence tentatively presumed that agglomeration or sintering of the reactant took place between cycles during different days. The effect of agglomeration can be mitigated by grinding or reactivating the reactants.

For the ball-milled reactant, the poor performance can be explained by the incomplete reaction of Fe₂O₃, MnCO₃ and Na₂CO₃. Although some MnFe₂O₄ can be recovered after one



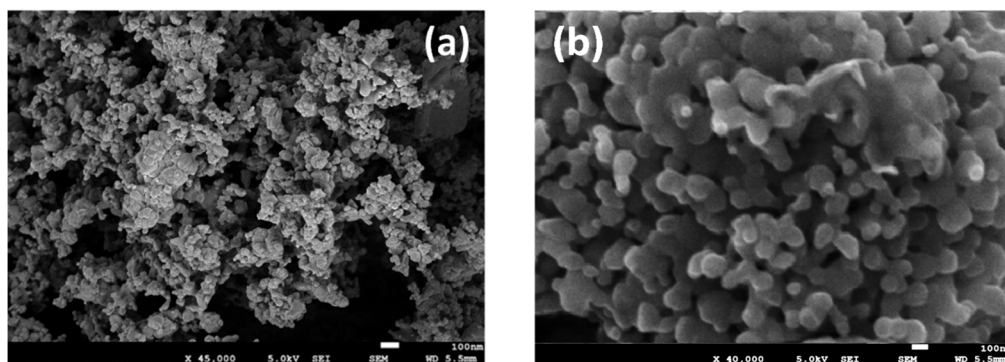


Fig. 5 MnFe_2O_4 and olivine, after water splitting at 700 °C (a) ball milled; (b) coprecipitation.

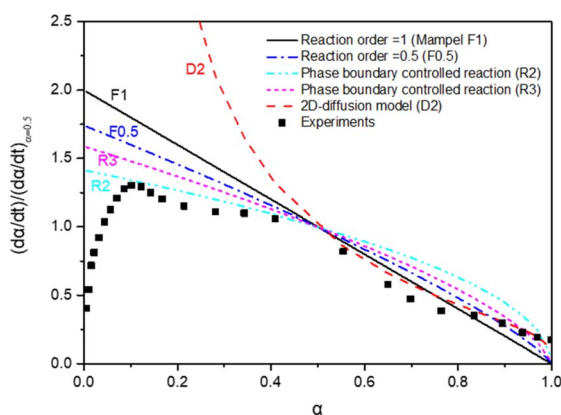


Fig. 6 Kinetic models applied to the experimental results: α is the reaction fraction $\left(\frac{C_{\text{H}_2} \text{ at time } t}{\text{theoretical total H}_2} \right)$.

cycle, some iron is segregated and forms Fe_2O_3 phase. The Fe_2O_3 phase will react with Na_2CO_3 and form NaFeO_2 , thus causing a decrease in H_2 productivity between the first and second cycle due to difficulties in regeneration step. The unreacted Fe_2O_3 when synthesizing the ball-milled catalysts also exacerbated the decrease in H_2 yield between cycles compared to the coprecipitated ones.

The solid-solid contact and ions transportation are equally important for the regeneration step. According to Chen *et al.*,⁴³ smaller particle size with moderate crystallinity is beneficial to maintain its structure stability and can lead to a better ionic transport within the crystals than through the grain boundaries in order to finish the whole cycle. This also counts for the difference in H_2 yield between the first two cycles. The particles smaller than 30 nm are good for the contact with Na_2CO_3 to release H_2 in the first step, but hamper removing Na^+ from the lamellar $\text{Na}(\text{Mn}_{1/3}\text{Fe}_{2/3})\text{O}_2$ oxide considering the low layered structure stability.

The cyclic O_2/H_2 production was however achieved. Fig. 5 compares typical SEM images of the ball-milled and coprecipitation reactants after water splitting at 700 °C. The particles synthesized by ball-milling (Fig. 5a) formed agglomerates consisting of 0.1 μm dense grains. The co-precipitation reactant

maintains a size in the order of 50–150 nm (Fig. 5b). The porous morphology of the MnFe_2O_4 reactants is maintained even after calcination and reaction at 700 °C.

The time-dependent conversion allows the study of the reaction kinetics. To evaluate appropriate kinetic models,²⁶ data were transformed in terms of reaction progress, α . These data were normalized against the conversion at $\alpha = 50\%$. All data transformations revealed similar fitting profiles, as illustrated in Fig. 6.

The MnFe_2O_4 water splitting is controlled by different reaction models. A 2D geometrical-contracting model (R2) seems appropriate when $\alpha < 0.5$. For $\alpha > 0.5$, a first-order (F1) or 2D diffusion model (D2) seems more appropriate. The morphology of Mn seems essential to the H_2 generation reaction kinetics. This is also the case for MnO_x -based water splitting reactions.⁴³ An additional kinetics study was carried out based upon the TGA results, as reported below.

3.2 The MnFe_2O_4 system studied in multicycle TGA experiments

TGA experiments provide an easy follow-up of the reaction progress through the weight changes of the sample under scrutiny. With a stoichiometric weight loss during the oxidation step of 83.18% (including the added O from H_2O), this

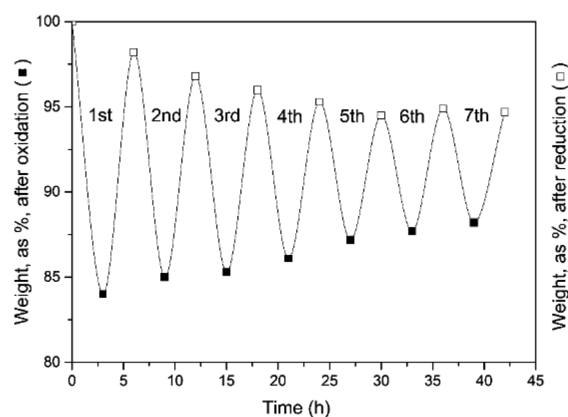


Fig. 7 Weight evolution of the oxidation (3 h) and reduction cycles (3 h). Seven cycles were examined.



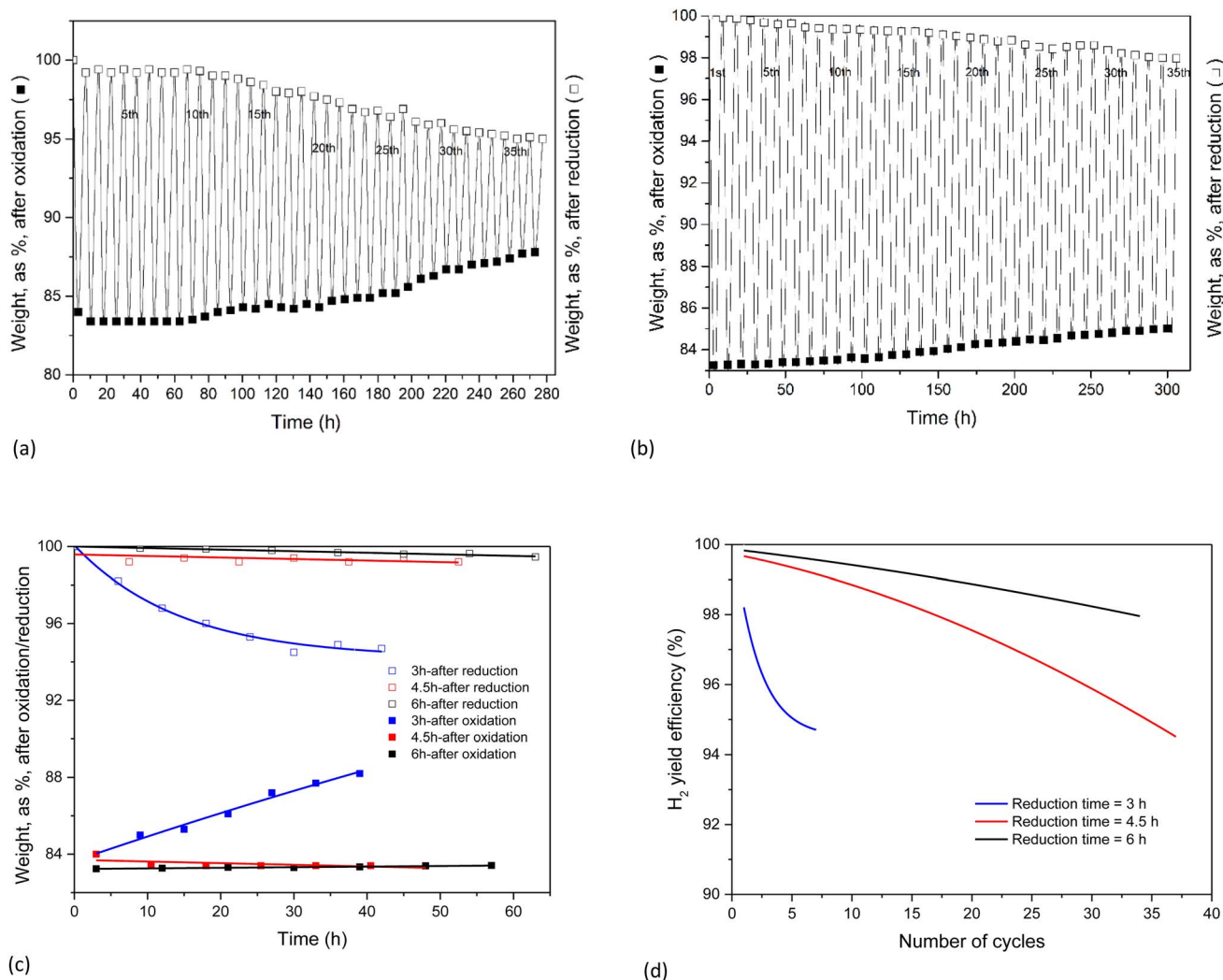


Fig. 8 Cyclic experiments at different reduction cycle times. (a) Weight evolution of, respectively the oxidation (3 h) and reduction cycles (4.5 h). (b) Weight evolution of, respectively the oxidation (3 h) and reduction cycles (6 h). (c) The first 7 cycles at a fixed oxidation time (3 h) but variable reduction time (3, 4.5, and 6 h, respectively). (d) H₂ yield efficiency.

theoretical limit forms the reference value for all the results. These results are illustrated in the following figures, obtained for a fixed oxidation duration of 3 h, but a variable reduction cycle of 3, 4.5 and 6 h.

The experimental results of Fig. 7 clearly demonstrate that a 3 hour CO₂-induced reduction cycle is insufficient to restore the initial activity of the reactant. The reaction was hence stopped after 7 cycles. This finding confirms the electrically heated furnace results.

Table 3 Reaction rate constant of the oxidation step

Reaction rate constant (min ⁻¹)	Reduction time		
	3 h	4.5 h	6 h
Cycle 1	2.23×10^{-2}	3.17×10^{-2}	3.54×10^{-2}
Cycle 6	1.68×10^{-2}	2.72×10^{-2}	3.07×10^{-2}
Cycle 30	—	1.77×10^{-2}	2.22×10^{-2}

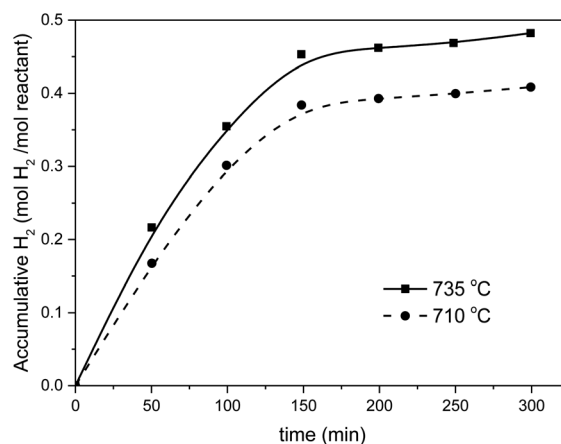


Fig. 9 Solar H₂ production using the MnFe₂O₄/Na₂CO₃ cycle.

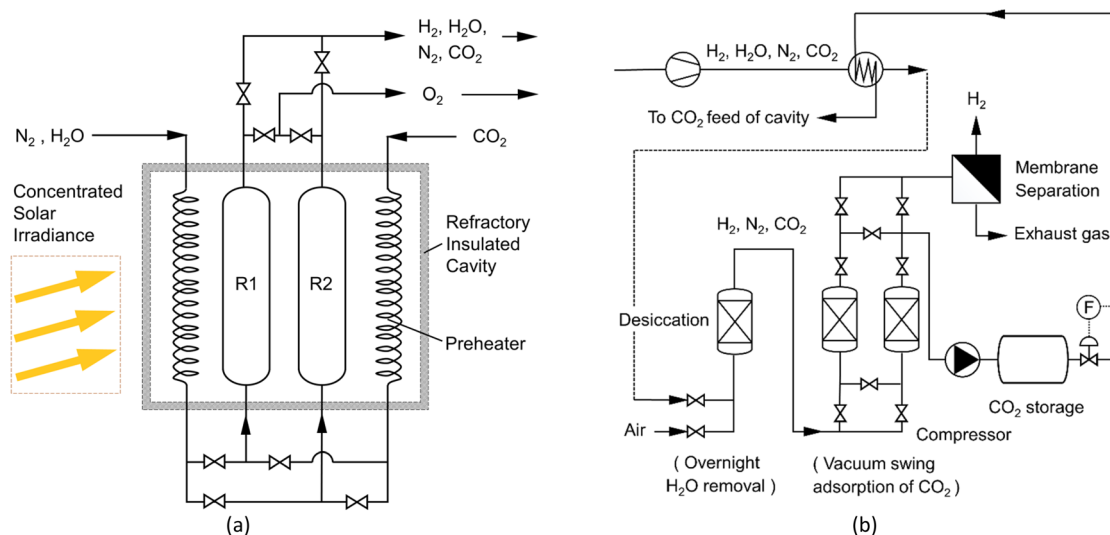


Fig. 10 (a) Installing redox-reactors in the cavity; (b) purification cycles of H_2 . Cycle 1 oxidation (R1) progresses in parallel with the reduction cycle (R2) by appropriate valve settings. The next solar day, the functions of (R1) and (R2) are reversed.

Long-duration, multicycle experiments with reduction cycles of 4.5 and 6 h were performed, and the results (Fig. 8a and b) clearly demonstrated that a longer reduction cycle enhances the recyclability.

All results for the first 7 cycles (with different total cycle times) are represented in Fig. 8c.

Finally, the system H_2 efficiency is calculated as follows, and presented in Fig. 8d:

$$\eta_{\text{H}_2} (\%) = \frac{1 - \text{experimental weight after oxidation cycle}}{1 - \text{theoretical weight after oxidation}} \times 100$$

The efficiency of the combined (3 + 6) h cycles exceeded 98% after 33 cycles, again 95% only for the (3 + 4.5) h operation. It is hence recommended to combine a 3 h oxidation (H_2 release) with 6 h of reduction (O_2 release).

The TGA results enable the calculation of the apparent reaction rate constant (k). Since $\alpha \gg 0.5$, first-order kinetics can be applied. Table 3 summarizes the results for the given numbers of cycles and a specific reduction time (3, 4.5, and 6 h).

The reaction rate decreases with the number of cycles for a given duration of the reduction, corresponding to a progressive but limited loss of activity of the solid reactant. The rate constant however increases when the reduction time is increased. It is hence important to either extend the oxidation cycle beyond 3 h as cycling proceeds, or to re-activate the reactants more frequently by calcination and/or re-milling after a certain number of cycles.

The reaction mechanism of the water splitting has been investigated by several researchers, both in electrocatalytic and thermochemical routes. Zhou *et al.*^{46,47} studied the electrocatalytic reduction reaction (ORR), also called the hydrogen evolution reaction (HER), and oxygen evolution reaction (OER). Four elementary reaction steps are proposed, including OOH

and OH species. Spin-polarized DFT simulations demonstrate that the reaction step are reported to be slow.⁴⁸ These 4 steps were confirmed by computational screenings.⁴⁹ The thermochemical mechanism was assessed by Chen *et al.*⁴³ and Angotzi *et al.*⁵⁰ with special emphasis on the slow oxygen release (over 3 h) of the $\text{Na}(\text{Mn}_{1/3}\text{Fe}_{2/3})\text{O}_2$ intermediate. In the hydrogen release step, lamellar $\text{Na}(\text{Mn}_{1/3}\text{Fe}_{2/3})\text{O}_2$ oxide is formed by intercalating a Na^+ layer into two adjacent oxygen interspaces.⁵¹ Both intercalation of Na^+ into adjacent oxygen interspaces and the Jahn–Teller effect lead to a more stable $\text{Na}(\text{Mn}_{1/3}\text{Fe}_{2/3})\text{O}_2$ oxide. The full mechanism is presented in Chen *et al.*⁴³

3.3 Preliminary results of the MnFe_2O_4 system in the solar reactor

Preliminary tests were performed with a solar reactor, during the first oxidation cycle. The temperature of the front reactor wall in the cavity was kept below 900 °C (thermal strength limits of the Ni/Cr construction alloy). The slow reaction of the heliostat focusing leads to temperature variations in the bed between 700 and 750 °C. Average values of 710 °C and 735 °C were maintained over a period of 5 h. The solar H_2 production was very good and the theoretical H_2 yield of 0.5 mol mol^{−1} was nearly achieved at 735 °C, as shown in Fig. 9.

Further to the TGA observations and prior to conducting additional cycling experiments that are now taking place in the season of high direct normal irradiance (DNI), the rig was adapted to conduct oxidation and reduction steps in parallel in the single cavity, as illustrated in Fig. 10.

The H_2 upgrading will apply a sequence of steps, involving the removal of excess H_2O vapour; the removal of CO_2 by vacuum swing adsorption⁶ on appropriate adsorbents, *e.g.* zeolite, activated carbon, or others;^{52,53} a membrane permeation of H_2 , *e.g.* using a Matrimid 5218⁵⁴ or P84 membranes⁵⁵ with over 95% H_2 purity and 96% recovery. The use of sintered metal fibre filters at the reactor exhausts limits the loss of reactant.⁵⁶



4. Recyclability

Results confirmed that the H_2 production efficiency can be $\gg 95\%$ provided oxidation and reduction cycles are properly performed. A conservative 95% efficiency can be accepted, as obtained both in TGA and in the solar receiver. For 0.5 mol of H_2 per mol reactant, a maximum of ~ 0.41 wt% H_2 can be generated. Coprecipitated $MnFe_2O_4$ is purchased at approximately 500 € per ton. The ball-milled reactant can be approximately 30% cheaper.

The heating costs are not considered since the system is supposed to use excess photovoltaic or wind turbine electricity, or to operate on concentrated solar heat. Heat recovery will be maximized by good heat management. The number of required cycles (N_c), to break even can be calculated for a proposed selling cost of H_2 , since:

$$N_c =$$

$$\frac{\text{cost of main reactant (€)}}{H_2 \text{ cost (€) wt } H_2 \text{ generated per unit amount of main reactant}}$$

CO_2 should be separated, stored, and used in the reverse reaction. It is also proposed to use membrane modules to produce very pure H_2 .^{54,55}

The solar energy balance is currently being assessed. At 4 € per kg H_2 and 500 € per ton of reactants, 30 cycles should be realized to break even. To reach 1 € per kg H_2 , 120 cycles should be achieved, and this is presumed possible in view of the obtained cycling results. If the cheaper ball-milled reactant could be improved, the number of required cycles to break even will be reduced.

5. Conclusions

This research demonstrated the high potential of water splitting by the $MnFe_2O_4/Na_2CO_3$ redox pair. Coprecipitated $MnFe_2O_4$ offers approximately 30% higher H_2 productivity than its cheaper ball-milled equivalent. Experiments in an electrically-heated reactor were used to study the fundamentals, possible shortcomings, and kinetics. TGA experiments demonstrated the long-duration and multicycle (up to 37 cycles) potential. Preliminary solar reactor experiments confirm the $>95\%$ H_2 efficiency: these solar experiments are ongoing in a joint oxidation/reduction parallel reactor, installed in the solar cavity. The CO_2 reaction time should be approximately 6 h to achieve a good reduction. Tentative cost calculations showed a break-even operation for 30 consecutive cycles at H_2 prices of 4 € per kg H_2 . At least 120 cycles before reactant regeneration will reduce the H_2 production cost to ~ 1 € per kg H_2 . This implies the use of a cheap energy supply, and the complete reuse of CO_2 in the reduction reaction. The results certainly foster a further improvement of the system.

Author contributions

R. Dewil and J. Baeyens supervised the research. Y. Deng, S. Li, and M. Yang designed and performed the experiments. L.

Appels and H. Zhang participated in the data analyses and discussions. Y. Deng, R. Dewil, and J. Baeyens wrote the paper. All authors discussed the results and commented on the manuscript.

Conflicts of interest

There are no conflicts to declare.

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Notes and references

- 1 J. Liu, S. Li, R. Dewil, M. Vanierschot, J. Baeyens and Y. Deng, *Sustainability*, 2022, **14**, 7597.
- 2 W. Antweiler, *What role does hydrogen have in the future of electric mobility?*, <https://wernerantweiler.ca/blog.php?item=2020-09-28>.
- 3 S. Li, Q. Kang, J. Baeyens, H. L. Zhang and Y. M. Deng, *IOP Conf. Ser.: Earth Environ. Sci.*, 2020, **544**, 012011.
- 4 BP p.l.c., *Hydroelectricity*, <https://www.bp.com/en/global/corporate/energy-economics/statistical-review-of-world-energy/hydroelectricity.html>.
- 5 Y. Deng, R. Dewil, L. Appels, S. Li, J. Baeyens, J. Degreè and G. Wang, *Renewable Energy*, 2021, **170**, 800–810.
- 6 C. Chao, Y. Deng, R. Dewil, J. Baeyens and X. Fan, *Renewable Sustainable Energy Rev.*, 2021, **138**, 110490.
- 7 Z. Li, H. Zhang, R. Dewil, Y. Deng and S. Li, *IOP Conf. Ser.: Earth Environ. Sci.*, 2022, **952**, 012007.
- 8 C.-J. Winter and J. Nitsch, *Hydrogen as an energy carrier: technologies, systems, economy*, Springer Science & Business Media, 2012.
- 9 S. Baykara, *Int. J. Hydrogen Energy*, 2004, **29**, 1451–1458.
- 10 J. E. Funk, *Int. J. Hydrogen Energy*, 2001, **26**, 185–190.
- 11 Y. Tamaura and M. Tahata, *Nature*, 1990, **346**, 255–256.
- 12 T. Sano, N. Hasegawa, M. Tsuji and Y. Tamaura, *J. Mater. Chem.*, 1996, **6**, 605–609.
- 13 T. Sano, T. Togawa, M. Kojima, M. Tsuji and Y. Tamaura, *Energy*, 1996, **21**, 377–384.
- 14 M. Inoue, N. Hasegawa, R. Uehara, N. Gokon, H. Kaneko and Y. Tamaura, *Sol. Energy*, 2004, **76**, 309–315.
- 15 S. R. Sabale, *Mater. Today: Proc.*, 2020, **23**, 139–146.
- 16 R. Asadi, H. Abdollahi, M. Gharabaghi and Z. Boroumand, *Adv. Powder Technol.*, 2020, **31**, 1480–1489.
- 17 N. Akhlaghi, G. Najafpour-Darzi, A. Barras, M. Mohammadi, R. Boukherroub and S. Szunerits, *Chem. Pap.*, 2021, **75**, 377–387.
- 18 I. Elahi, R. Zahira, K. Mehmood, A. Jamil and N. Amin, *Afr. J. Pure Appl. Chem.*, 2012, **6**, 1–5.
- 19 L. A. Kafshgari, M. Ghorbani and A. Azizi, *Part. Sci. Technol.*, 2019, **37**, 904–910.



- 20 G. Wang, Y. Zeng, F. Zhou, X. Chen, Y. Ma, L. Zheng, M. Li, Y. Sun, X. Liu, H. Liu and R. Yu, *J. Alloys Compd.*, 2020, **819**, 153044.
- 21 A. A. Ati, A. H. Abdalsalam and A. S. Hasan, *J. Mater. Sci.: Mater. Electron.*, 2021, **32**, 3019–3037.
- 22 J. Amighian, M. Mozaffari and B. Nasr, *Phys. Status Solidi C*, 2006, **3**, 3188–3192.
- 23 A. Hassan, W. Ding, M. A. Aslam, Y. Bian, Q. Liu and Z. Sheng, *J. Mater. Res. Technol.*, 2020, **9**, 12869–12879.
- 24 H. Yang, C. Zhang, X. Shi, H. Hu, X. Du, Y. Fang, Y. Ma, H. Wu and S. Yang, *Biomaterials*, 2010, **31**, 3667–3673.
- 25 G. Singh and S. Chandra, *Int. J. Hydrogen Energy*, 2018, **43**, 4058–4066.
- 26 R. H. Vignesh, K. V. Sankar, S. Amaresh, Y. S. Lee and R. K. Selvan, *Sens. Actuators, B*, 2015, **220**, 50–58.
- 27 J. Kwon, J.-H. Kim, S.-H. Kang, C.-J. Choi, J. A. Rajesh and K.-S. Ahn, *Appl. Surf. Sci.*, 2017, **413**, 83–91.
- 28 H. Tombuloglu, G. Tombuloglu, Y. Slimani, I. Ercan, H. Sozeri and A. Baykal, *Environ. Pollut.*, 2018, **243**, 872–881.
- 29 W. Zhang, X. Hou, Z. Lin, L. Yao, X. Wang, Y. Gao and S. Hu, *J. Mater. Sci.: Mater. Electron.*, 2015, **26**, 9535–9545.
- 30 K. Vamvakidis, M. Katsikini, D. Sakellari, E. C. Paloura, O. Kalogirou and C. Dendrinou-Samara, *Dalton Trans.*, 2014, **43**, 12754–12765.
- 31 Z. Liu, G. Chen, X. Li and X. Lu, *J. Alloys Compd.*, 2021, **856**, 158185.
- 32 M. Goodarz Naseri, E. Bin Saion, H. A. Ahangar, M. Hashim and A. H. Shaari, *J. Magn. Magn. Mater.*, 2011, **323**, 1745–1749.
- 33 X.-Y. Long, J.-Y. Li, D. Sheng and H.-Z. Lian, *Talanta*, 2017, **166**, 36–45.
- 34 F. T. L. Muniz, M. A. R. Miranda, C. Morilla dos Santos and J. M. Sasaki, *Acta Crystallogr., Sect. A: Found. Adv.*, 2016, **72**, 385–390.
- 35 F. Varsano, M. A. Murmura, B. Brunetti, F. Padella, A. La Barbera, C. Alvani and M. C. Annesini, *Int. J. Hydrogen Energy*, 2014, **39**, 20920–20929.
- 36 F. Padella, C. Alvani, A. La Barbera, G. Ennas, R. Liberatore and F. Varsano, *Mater. Chem. Phys.*, 2005, **90**, 172–177.
- 37 H. Kaneko, Y. Hosokawa, N. Gokon, N. Kojima, N. Hasegawa, M. Kitamura and Y. Tamaura, *J. Phys. Chem. Solids*, 2001, **62**, 1341–1347.
- 38 L. Seralessandri, M. Bellusci, F. Padella, A. Santini and F. Varsano, *Int. J. Hydrogen Energy*, 2009, **34**, 4546–4550.
- 39 M. A. Murmura, F. Varsano, F. Padella, A. La Barbera, C. Alvani and M. C. Annesini, *Ind. Eng. Chem. Res.*, 2014, **53**, 10310–10317.
- 40 F. Varsano, F. Padella, C. Alvani, M. Bellusci and A. La Barbera, *Int. J. Hydrogen Energy*, 2012, **37**, 11595–11601.
- 41 A. Mottana, R. Crespi and G. Liborio, *Simon & Schuster's guide to Rocks and Minerals*, Simon and Schuster, New York, 1991.
- 42 L. Y. Dolgikh, I. L. Stolyarchuk, L. A. Staraya, I. V. Vasylenko, Y. I. Pyatnitsky and P. E. Strizhak, *Catal. Petrochem.*, 2020, 1–10.
- 43 Z. Chen, Q. Jiang, J. Tong, M. Yang, Z. Jiang and C. Li, *Sol. Energy*, 2016, **129**, 236–243.
- 44 S. Li, H. Zhang, J. Nie, R. Dewil, J. Baeyens and Y. Deng, *Sustainability*, 2021, **13**, 8866.
- 45 C. C. Agrafiotis, C. Pagkoura, A. Zygogianni, G. Karagiannakis, M. Kostoglou and A. G. Konstandopoulos, *Int. J. Hydrogen Energy*, 2012, **37**, 8964–8980.
- 46 Y. Zhou, Y. Du, S. Xi and Z. J. Xu, *Electrocatalysis*, 2018, **9**, 287–292.
- 47 Y. Zhou, G. Gao, J. Kang, W. Chu and L.-W. Wang, *Nanoscale*, 2019, **11**, 18169–18175.
- 48 Y. Singh, S. Back and Y. Jung, *Phys. Chem. Chem. Phys.*, 2018, **20**, 21095–21104.
- 49 Y. Yang, J. Liu, F. Liu, Z. Wang and D. Wu, *J. Mater. Chem. A*, 2021, **9**, 2438–2447.
- 50 M. Sanna Angotzi, V. Mameli, C. Cara, D. Peddis, H. L. Xin, C. Sangregorio, M. L. Mercuri and C. Cannas, *Nanoscale Adv.*, 2021, **3**, 1612–1623.
- 51 C. Alvani, G. Ennas, A. Labarbera, G. Marongiu, F. Padella and F. Varsano, *Int. J. Hydrogen Energy*, 2005, **30**, 1407–1411.
- 52 Y. Deng, J. P. K. Seville, S. D. Bell, A. Ingram, H. Zhang, N. Sweygers, R. Dewil, J. Baeyens and L. Appels, *Sep. Purif. Rev.*, 2021, 1–12.
- 53 J. P. K. Seville, Y. Deng, S. Dawn Bell, R. Dewil, L. Appels, R. Ansart, T. Leadbeater, D. Parker, H. Zhang, A. Ingram and J. Baeyens, *Chem. Eng. J.*, 2021, **404**, 126507.
- 54 R. Castro-Muñoz, V. Martín-Gil, M. Z. Ahmad and V. Fila, *Chem. Eng. Commun.*, 2018, **205**, 161–196.
- 55 G. Li, Z. Si, S. Yang, T. Xue, J. Baeyens and P. Qin, *Chem. Eng. Sci.*, 2022, **253**, 117588.
- 56 S. Li, J. Baeyens, R. Dewil, L. Appels, H. Zhang and Y. Deng, *Renewable Sustainable Energy Rev.*, 2021, **139**, 110713.

