


Cite this: *RSC Adv.*, 2022, 12, 29010

# Exploration of the interesting photovoltaic behavior of the fused benzothiophene dioxide moiety as a core donor with modification in acceptors for high-efficacy organic solar cells†

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Non-fullerene-based chromophores with end-capped acceptor modification used in organic solar cells (OSCs) have proven to offer improved performance. Therefore, eight unique benzothiophene-based molecules (D2–D9) were designed by the end-capped acceptor manipulation of a reference molecule (R1). Density functional theory (DFT) and time-dependent density functional theory (TD-DFT) calculations at the B3LYP level were performed to investigate various parameters such as the optical properties, frontier molecular orbitals (FMOs), transition density matrix (TDM), binding energy, density of states (DOS), open-circuit voltage ( $V_{oc}$ ), and reorganization energies of electrons ( $\lambda_e$ ) and holes ( $\lambda_h$ ), to better understand the optoelectronic properties of the newly designed compounds. All the derivatives had broader absorption spectra of 737.562–700.555 nm with a reduced energy gap of 2.132–1.851 eV compared to those of the reference (719.082 nm), except for D8 and D9. A comparable value of  $V_{oc}$  and lower reorganization energies were found in the derivatives compared to those of R1. Within the studied compounds, D3 was predicted to be a good optoelectronic material for environmentally friendly organic solar cells (EFOSCs) because of its superior optoelectronic capabilities, low-energy band gap (1.851 eV), highest  $\lambda_{max}$  values of 794.516 and 744.784 nm in chloroform and the gas phase, respectively, and lowest transition energy (1.561 eV) than those of the reference and the other derivatives. Subsequently, end-capped acceptor modification was proven to be an effective method to achieve desired optoelectronic characteristics.

Received 19th July 2022

Accepted 5th September 2022

DOI: 10.1039/d2ra04473k

rsc.li/rsc-advances

## 1 Introduction

Over the last few years, the performance of organic solar cells has dramatically improved, particularly due to the synthesis of small compounds such as non-fullerene acceptors (NFAs) that can cost-effectively produce an impressive photovoltaic response.<sup>1,2</sup> Organic compounds such as non-fullerene electron acceptors have improved solar light-emitting performance and energy efficiency.<sup>3</sup> Currently, organic solar cells based on NFAs have a power conversion efficiency (PCE) of 18%. The applications of NFAs with an acceptor–donor–acceptor (A–D–A) structure have been receiving increasing attention due to their charge mobilities, long-range absorption, and fine-tuned energy levels.<sup>4</sup> The most powerful NFAs have A–D–A structures, with a fused electron-donating core and two electron-accepting end groups. Therefore, a lot of researchers are working on acceptors other than the fullerene A–D–A structure with optimized charge dispersions, absorption, and energy levels<sup>5</sup> to improve the PCE values.<sup>6</sup> For quite a long time, silicon was considered the most suitable material for use as a semiconductor in photovoltaic

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† Electronic supplementary information (ESI) available: 2D structures with their IUCAP names, Cartesian co-ordinates and UV-vis data (wave lengths, excitation energies and oscillator strengths) were calculated using B3LYP/6-31G (d,p) for the studied compounds and are represented in the ESI data file. See DOI: <https://doi.org/10.1039/d2ra04473k>



devices. The key factors behind the prominent position of silicon in solar cells during the last 60 years of development include its extensive stability, large natural abundance,<sup>7</sup> eco-friendly nature and high PCE of about 46%. However, a limiting factor in these types of solar cells is the increase in cost per watt with the increase in efficiency along with a low light-absorption rate.<sup>8</sup> Compared with traditional silicon solar cells, dye-sensitized solar cells (DSSCs) have several merits such as light weight, low cost, easy fabrication and flexibility.<sup>9</sup> This might be due to the narrow absorption spectrum range and quick recombination of electron-hole pairs on the surface of the photoanode.<sup>10</sup> In recent years, organic solar cells (OSCs) have proved to be an efficient and successful replacement of traditional silicon and DSSCs in terms of energy generation. OSCs have cemented their place in foldable and flexible applications because of their flexibility and light weight. OSCs as an attractive renewable energy platform offer advantages in terms of their flexibility, strength, tunable energy levels, wide area for fabrication and low cost.<sup>11</sup> Fullerene-based OSCs offer a notable PCE of 12% in bulk heterojunctions (BHJs), affording a sufficient short-circuit current ( $J_{sc}$ ) and open-circuit voltage ( $V_{oc}$ ).<sup>12</sup> OSCs in combination with fullerene acceptors can achieve high isotropic charge mobility<sup>13</sup> and low reorganization energy. Despite their good performance, limitations such their high cost, non-tunable capacity of the lowest unoccupied molecular orbital (LUMO) energy, reduced absorption of sunlight in the visible region, low  $V_{oc}$ , high band gap and poor morphological reliability make fullerene acceptors less efficient.<sup>14</sup> To overcome these downsides, scientists are working on designing organic photovoltaic materials based on non-fullerene acceptors (NFAs). These non-fullerene organic photovoltaic (NF-OPV) materials are good replacements for their fullerene counterparts due to their acclaimed properties such as extended light-absorption ability, low voltage loss, easy molecular engineering for fabrication, solubility, tunable energy levels, and high morphological and photochemical stability.<sup>15</sup> Most of the organic non-fullerene acceptor dyes retain the acceptor-donor-acceptor (A-D-A) backbone architecture. The A-D-A combination comprises a central electron-donating core unit blended with two sideways end-capped acceptor moieties. The most prominent reported example of the A-D-A architecture is ITIC,<sup>16</sup> which shows promising optoelectronic properties such as a small band gap and long absorption range with a PCE of 16%.<sup>17</sup> The most effective strategy to enhance the photovoltaic properties of NF-OSCs is to narrow the HOMO-LUMO energy gap *via* structural alteration and functional-group modification. Selecting proper electron-donor and -acceptor components can significantly reduce the HOMO-LUMO energy gap, increase the PCE and improve the photovoltaic properties of non-fullerene-based OSCs. Appropriate arrangement of the acceptor, donor and acceptor plays a vital role in designing potential organic compound dyes. Keeping in mind the above discussion, we utilized **R1**, an NF-SMA with benzothiophene dioxide (BO) terminal groups attached to phenyl side moieties, as a reference molecule.<sup>18</sup> A series of eight new fullerene-free acceptor chromophores has been developed from **R1**. End-capped acceptor modifications were brought about in the reference

chromophore. The electronic, photophysical, and photovoltaic materials of the predicted molecules (**D2–D9**) and their charge transfer in comparison to that of **R1** have been illustrated using TDM, FMO, dipole moment, DOS, reorganization energy,  $V_{oc}$  and absorption maximum analyses. It is expected that these newly designed derivatives will play a key role in the progress of high-efficacy OSC materials.

## 2 Results and discussion

### 2.1 Molecular chemistry

This work used the DFT technique to study the optoelectronic properties of eight novel fullerene-free acceptor molecules (**D2–D9**) designed *via* structural manipulation at the terminal acceptors. For this, **ITBCR**<sup>19</sup> was selected as the parent molecule, comprising a central benzo[*b*]thiophene-3(2*H*)-one 1,1-dioxide core unit linked by (*Z*)-2-(2-ethylidene-1,1-dioxidobenzo[*b*]thiophen-3(2*H*)-ylidene)malononitrile, which acts as a donor (D) moiety along with terminal electron-withdrawing acceptor (A) units. To eliminate steric hindrance and reduce the computational cost brought about by the long alkyl chains in **ITBCR**, the alkyl group  $C_6H_{13}$  was replaced with a methyl ( $-CH_3$ ) group, as shown in Fig. 1, and its name was changed from **ITBCR** to **R1** (Scheme 1).

We changed the terminal acceptors of **R1** with various well-known end-capped acceptors to explore and boost the optoelectronic and photophysical properties of OSCs. The structures of **R1** and its derivatives (**D2–D9**) in optimized form are depicted in Fig. 2.

### 2.2 Frontier molecular orbital (FMO) analysis

The electrical and optical properties of **R1** and **D2–D9** were studied *via* FMO analysis.<sup>20</sup> The HOMO is generally an electron donor, whereas the LUMO is considered to be an electron acceptor.  $E_{LUMO} - E_{HOMO}$  is known as the band gap ( $E_g$ ).<sup>21</sup> FMO investigations of **R1** and the designed chromophores (**D2–D9**) were performed using the B3LYP/6-31G(d,p) level of DFT, and the FMO orbitals are shown in Fig. 3. Large  $E_{HOMO} - E_{LUMO}$  energy gap molecules are termed as hard molecules with low chemical reactivity and great stability. However, small  $E_{LUMO} - E_{HOMO}$  energy gaps are associated with soft molecules possessing high reactivity, low stability and great polarizability.<sup>22,23</sup> The energy gap and energies of the HOMO ( $E_{HOMO}$ ) and LUMO ( $E_{LUMO}$ ) of **R1** and **D2–D9** were calculated and the results are shown in Table 1.

The calculated values of  $E_{HOMO}$  and  $E_{LUMO}$  for **R1** are found to be  $-5.574$  and  $-3.571$  eV, respectively, with an energy gap of 2.003 eV. Meanwhile, the experimental values for the HOMO/LUMO have been determined to be  $-5.61/-4.13$  eV with a 1.48 eV energy gap. A close harmony is thus seen between the DFT and experimental values, which confirms the suitable selection of the functional used for computational analysis.<sup>19</sup> Similarly, the  $E_{HOMO}$  energies are determined to be  $-5.476$ ,  $-5.591$ ,  $-5.516$ ,  $-5.580$ ,  $-5.489$ ,  $-5.577$ ,  $-5.427$  and  $-5.400$  eV, while the  $E_{LUMO}$  values are  $-3.476$ ,  $-3.740$ ,  $-3.549$ ,  $-3.689$ ,  $-3.519$ ,  $-3.685$ ,  $-3.295$  and  $-3.297$  eV for **D2–D9**, respectively.

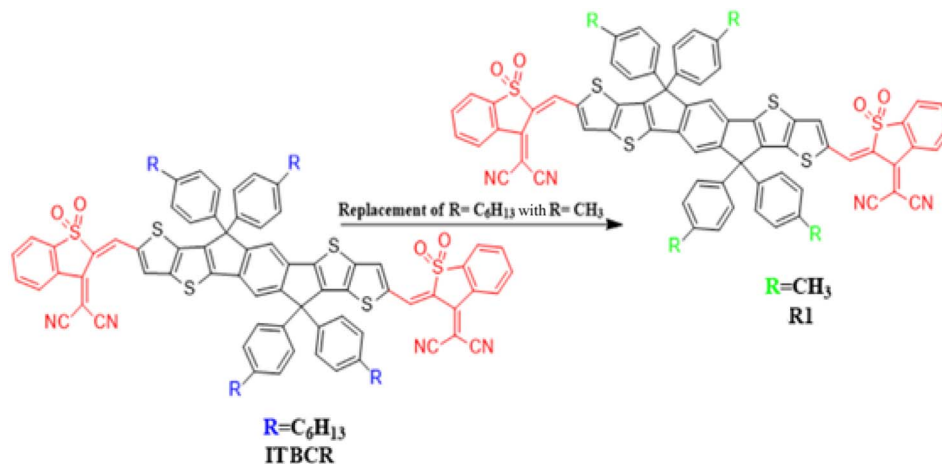
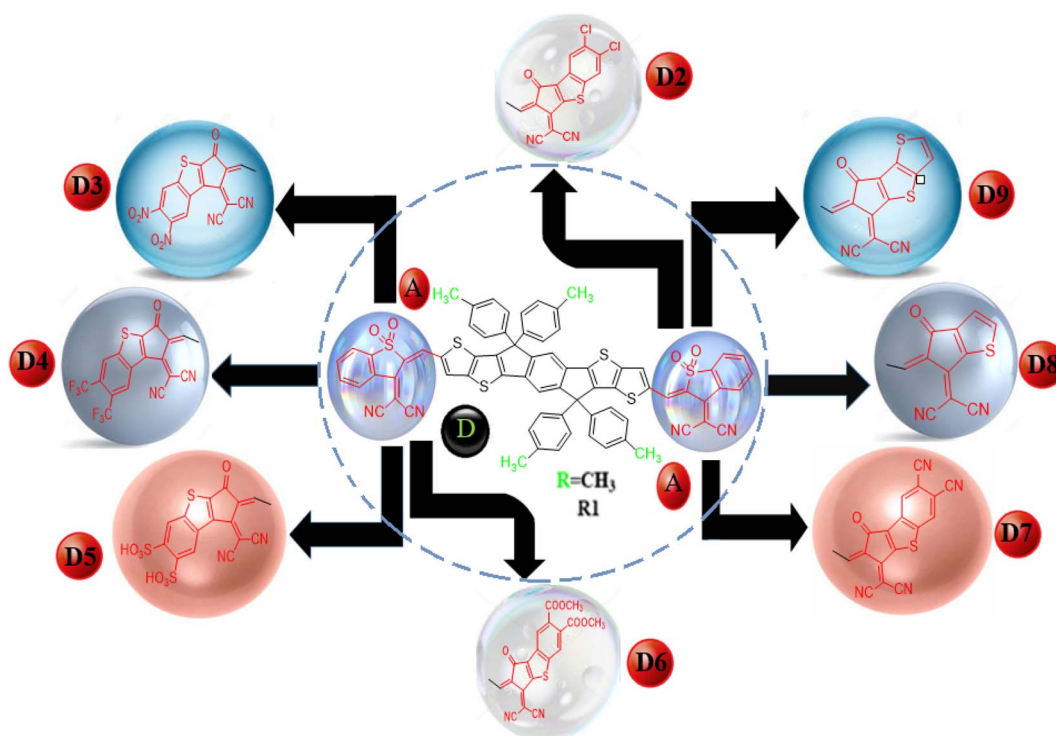


Fig. 1 Modification of ITBCR into R1 via the substitution of a  $-CH_3$  group.

The most important factor for describing the charge transport and optoelectronic properties in molecules is  $\Delta E$ , and the energy gaps of **R1** and **D2–D9** are found to be 2.003, 2.000, 1.851, 1.967, 1.891, 1.97, 1.892, 2.132 and 2.103 eV, respectively (Table 1). The  $E_g$  has been found to decrease in all the designed compounds (**D3–D7**) except for **D8** and **D9**, which exhibited band gaps a little bit larger than that of **R1**. In **D2**, a reduction in the band gap (2.000 eV) is found due to the addition of a thiophene ring and chloro ( $-Cl$ ) groups at the acceptor moiety, which may enhance the electron-withdrawing effect along with the resonance. The band gap in **D3** (1.851 eV) is found to decrease because the  $-Cl$  groups are replaced with nitro ( $-NO_2$ )

groups at the acceptor unit. As  $-NO_2$  groups have a greater negative inductive ( $-I$ ) effect than  $-Cl$  ( $NO_2 > Cl$ ),<sup>24</sup> the energy gap of **D3** is reduced compared to that of **D2**. In the case of compound **D4**, an increase in the band gap (1.967 eV) is found compared to that of **D3**, as the  $-NO_2$  groups at the acceptor are replaced with trifluoromethyl ( $-CF_3$ ) groups. This might be due to  $-CF_3$  exhibiting a lower electron-withdrawing ability than the  $-NO_2$  group.<sup>25</sup> Further decline in the band gap is found in **D5** (1.891 eV), in which the  $-CF_3$  groups at the acceptor unit are replaced with sulfonic acid ( $-SO_3H$ ) groups. Moreover, with the replacement of the  $-SO_3H$  group with the acetic acid ( $-CH_3-COOH$ ) group at the acceptor moiety, the  $\Delta E$  of **D6** (1.970 eV) is



Scheme 1 Schematic of the non-fullerene acceptor units.



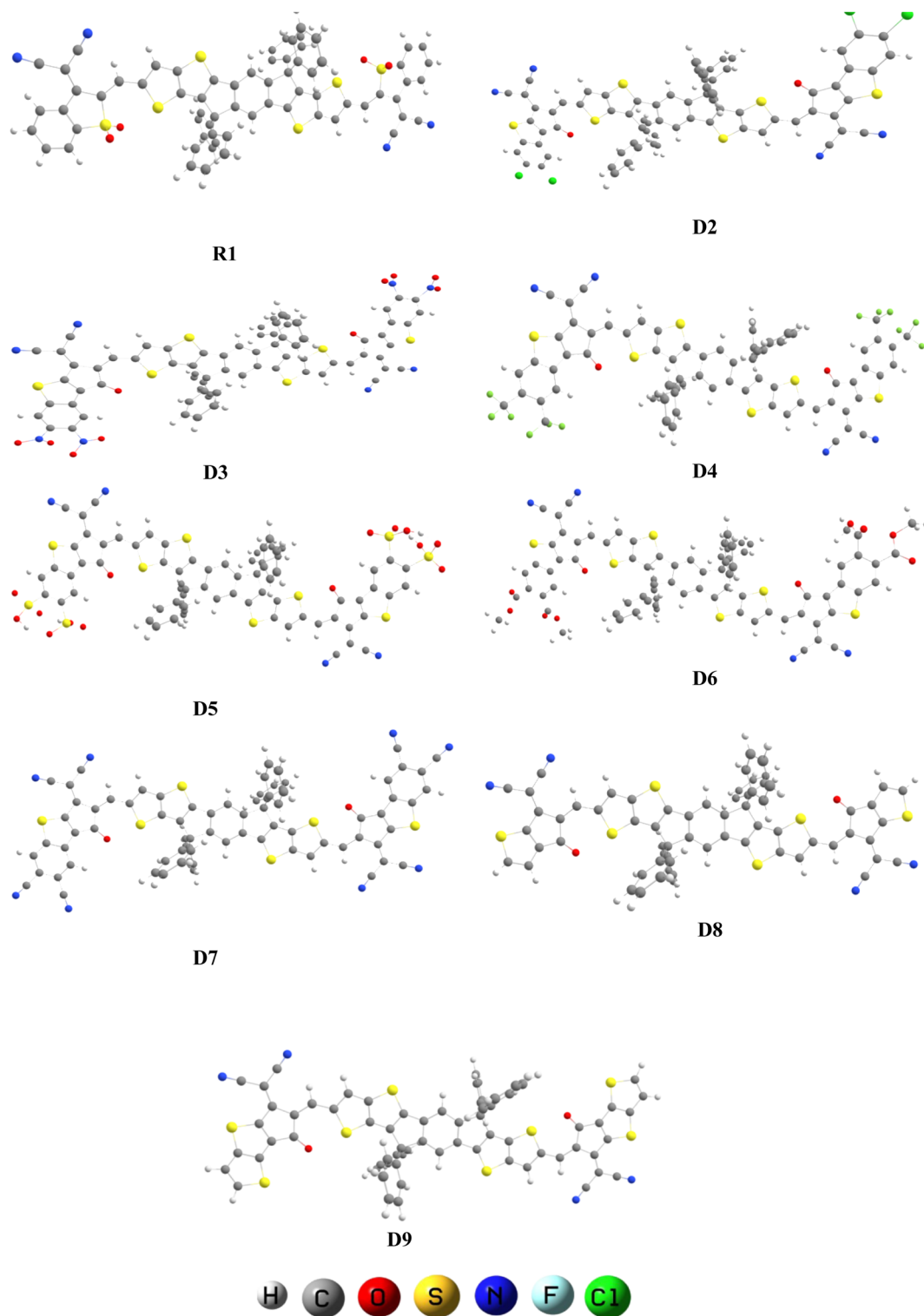


Fig. 2 Optimized geometries of R1 and D2–D9.

found to increase. This increment in the energy gap might be due to the weaker electron-withdrawing  $-\text{CH}_3\text{COOH}$  group (compared to the  $-\text{SO}_3\text{H}$  group). The band gap of **D7** (1.892 eV) is less than that of **D6** (1.970 eV) because of the replacement of  $-\text{CH}_3\text{COOH}$  with  $-\text{CN}$  groups at the acceptor moiety. This

reduction in the energy gap is due to the  $-\text{I}$  effect of the electron-withdrawing  $-\text{CN}$  groups. In compound **D8**, the band gap (2.132 eV) is observed to increase compared to that of **D7** (1.892 eV) because of the removal of one fragment (2-ethylidene-3-methylene-succinonitrile) from the acceptor unit. Owing to



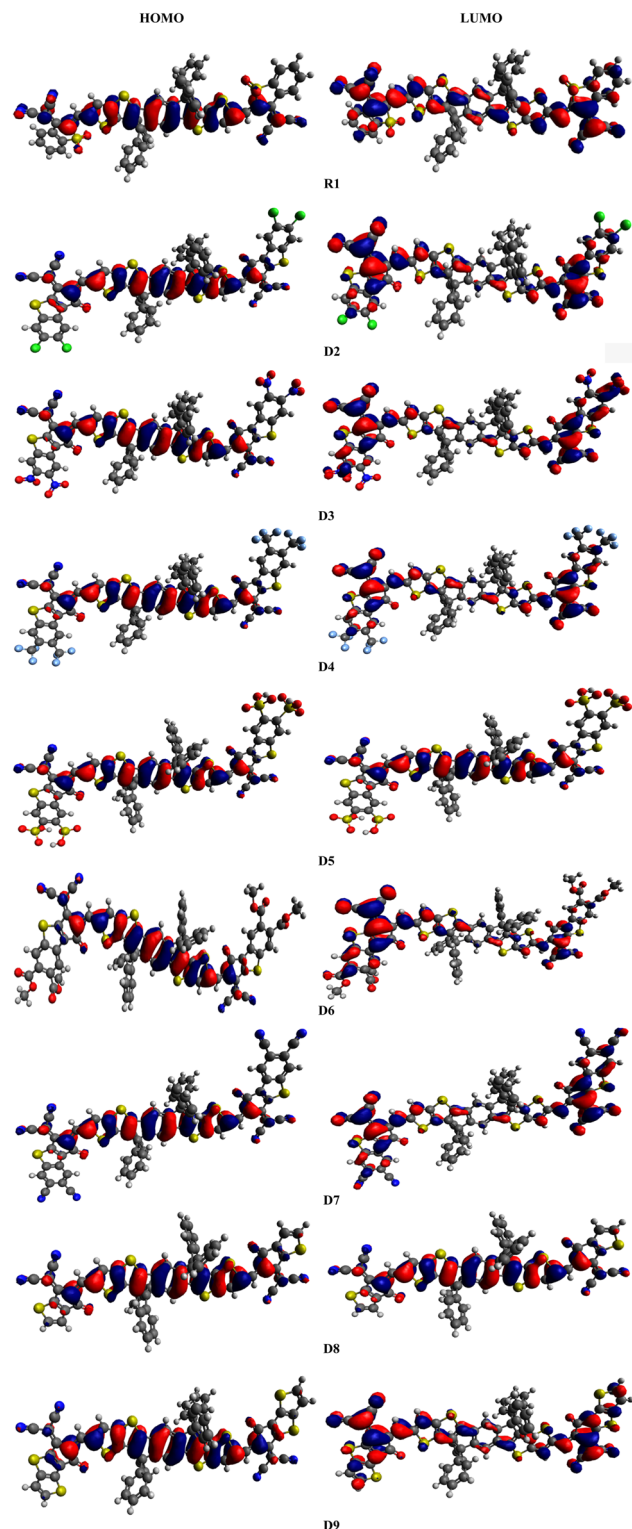


Fig. 3 Frontier molecular orbital diagrams of R1 and D2–D9.

the removal of this fragment, the conjugation of the system is reduced and the  $E_g$  was enhanced. Finally, the energy gap of **D9** (2.103 eV) is observed to decrease compared to that of **D8** (2.132 eV) because of the replacement of a benzene ring with a thiophene ring into the system at the terminal acceptor. Because of

Table 1 Calculated  $E_{\text{HOMO}}$ ,  $E_{\text{LUMO}}$  and energy gap ( $\Delta E$ ) of compounds R1 and D2–D9<sup>a</sup>

Compounds	$E_{\text{HOMO}}$	$E_{\text{LUMO}}$	Band gap
<b>R1</b>	−5.574	−3.571	2.003
<b>D2</b>	−5.476	−3.476	2.000
<b>D3</b>	−5.591	−3.740	1.851
<b>D4</b>	−5.516	−3.549	1.967
<b>D5</b>	−5.580	−3.689	1.891
<b>D6</b>	−5.489	−3.519	1.970
<b>D7</b>	−5.577	−3.685	1.892
<b>D8</b>	−5.427	−3.295	2.132
<b>D9</b>	−5.400	−3.297	2.103

<sup>a</sup> Band gap =  $E_{\text{LUMO}} - E_{\text{HOMO}}$  (eV).

Table 2 Calculated energy ( $E$ ), wavelength ( $\lambda_{\text{max}}$ ), oscillation strength ( $f_{\text{os}}$ ) and nature of the molecular orbital (MO) contributions of R1 and D2–D9 in chloroform

Compound	$\lambda$ (nm)	$E$ (eV)	$f_{\text{os}}$	MO contributions
<b>R1</b>	719.082	1.724	2.739	H → L (99%)
<b>D2</b>	737.562	1.681	1.786	H → L (97%)
<b>D3</b>	794.516	1.561	1.663	H → L (98%)
<b>D4</b>	747.839	1.658	1.781	H → L (97%)
<b>D5</b>	778.257	1.593	1.738	H → L (98%)
<b>D6</b>	747.929	1.658	1.750	H → L (97%)
<b>D7</b>	778.795	1.592	1.711	H → L (97%)
<b>D8</b>	682.657	1.816	2.229	H → L (98%)
<b>D9</b>	700.555	1.770	1.930	H → L (96%)

Table 3 Calculated energy ( $E$ ), wavelength ( $\lambda_{\text{max}}$ ), oscillation strength ( $f_{\text{os}}$ ) and nature of the molecular orbital (MO) contributions of R1 and D2–D9 in the gas phase

Compound	$\lambda$ (nm)	$E$ (eV)	$f_{\text{os}}$	MO contributions
<b>R1</b>	673.242	1.842	2.375	H → L (100%)
<b>D2</b>	693.928	1.787	1.498	H → L (96%)
<b>D3</b>	744.784	1.665	1.343	H → L (97%)
<b>D4</b>	705.217	1.758	1.465	H → L (97%)
<b>D5</b>	738.749	1.678	1.387	H → L (97%)
<b>D6</b>	701.030	1.769	1.433	H → L (97%)
<b>D7</b>	734.504	1.688	1.374	H → L (97%)
<b>D8</b>	643.773	1.926	1.885	H → L (98%)
<b>D9</b>	661.955	1.873	1.618	H → L (96%)

this alteration, the conjugation in the molecule is reduced, resulting in an enhancement in the band gap. Overall, in **R1** and **D2–D9**, the  $\Delta E$  values are found to decrease in the order: **D8** > **D9** > **R1** > **D2** > **D6** > **D4** > **D7** > **D5** > **D3**. The electrical cloud transmission phenomenon is frequently described using the contour elements in FMO.<sup>26</sup> Pictographs of the molecular orbitals exhibiting the charge densities of the aforementioned compounds are presented in Fig. 3. Generally, the HOMO density lies over the central donor unit and a slight amount of it is over the peripheral acceptors; whereas for the LUMO, the electronic cloud is found predominantly over the end-capped acceptor moieties, and a little amount of it is concentrated



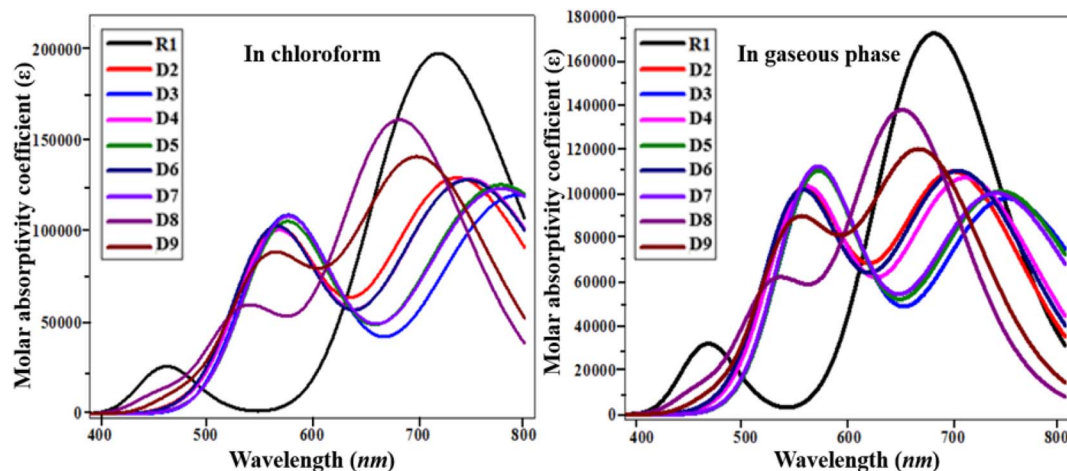


Fig. 4 UV-visible absorption spectra of the investigated molecules (R1 and D2–D9).

Table 4 Computed reorganization energies of the R1 and D2–D9 chromophores

Compounds	$\lambda_e$ (eV)	$\lambda_h$ (eV)
R1	−0.00002673	0.00018773
D2	−0.00000762	−0.33841872
D3	0.00003289	0.00009158
D4	−0.00003688	0.00005475
D5	0.00005186	0.00007521
D6	−0.00004563	0.00011363
D7	0.00002929	0.00008763
D8	0.00026435	0.00009166
D9	0.0001786	0.00009157

over the central donor, except for **D5** and **D8**, where the charge is located over the central part (Fig. 3).

### 2.3 Optical properties

To elucidate the working potential of optoelectronic devices, UV-vis spectral analysis can provide a clear understanding of the spectral and optoelectronic properties.<sup>27</sup> UV-vis spectroscopy is a useful technique for the determination of the nature of transitions and charge-transfer characteristics. The absorption spectra of the designed compounds **R1** and **D2–D9** were calculated using TD/DFT computation at the B3LYP/6-31G(d,p) functional level in chloroform and in the gaseous phase.<sup>28</sup> Various parameters such as the  $\lambda_{\max}$ , transition energy, oscillation strength ( $f_{\text{os}}$ ) and molecular orbital contributions of the studied chromophores have been presented in Tables 2 and 3. The simulated absorption spectra of the analyzed chromophores (**R1**, **D2–D9**) are portrayed in Fig. 4.

In chloroform solvent, the values of the maximum absorbed wavelength exist in the range of 682.657 to 794.516 nm. **D3** showed a maximum absorption wavelength of 794.516 nm, the furthest away from the experimental outcome for the parent chromophore (711 nm).<sup>29</sup> It also exhibited the lowest  $E$  among the reference and derivatives due to its strong electron-withdrawing nitro group in the acceptor unit, with an

excitation energy of 1.561 eV. A low excitation energy and low band gap lead to improved charge mobility with a high power conversion efficiency.<sup>30</sup> In chloroform, the maximum absorption peaks for the **D2–D9** molecules are 737.562, 794.516, 747.839, 778.257, 747.929, 778.795, 682.657 and 700.555 nm, respectively. The lowest value of  $\lambda_{\max}$  is in the case of **D8** (682.657 nm) due to the attachment of a weakly electron-withdrawing end-capped acceptor group. In the chloroform solvent, the decreasing order of the studied chromophores with respect to their  $\lambda_{\max}$  in chloroform is **D3** > **D7** > **D5** > **D6** > **D4** > **D2** > **R1** > **D9** > **D8**.

In the gaseous phase, the maximum absorbed wavelength of all the investigated chromophores also lies in the visible region. Herein, the calculated absorbed wavelength was examined in the range of 643.773–744.784 nm. The maximum absorbed wavelength of the reference chromophore (**R1**) is 673.242 nm. Except for **D8** and **D9**, all the investigated chromophores possess higher wavelengths than the reference molecule. Like in the solvent phase, **D3** also exhibited the highest  $\lambda_{\max}$  among the investigated molecules. A small decrease in the values of  $\lambda_{\max}$  of the studied chromophores in the gas phase is observed, which might be due to the solvent effect,<sup>31</sup> as shown in Table 3. In the gas phase, the maximum absorbed wavelength decreases in the order of **D3** > **D5** > **D7** > **D4** > **D6** > **D2** > **R1** > **D9** > **D8**. **D3** shows the smallest transition energy, the lowest energy gap, and the largest maximum value, which are favorable parameters for utilizing a material's optoelectronic properties in organic solar cell applications.<sup>32</sup> The above discussion concludes that red-shifted compounds with a low energy gap and a strong charge-transfer ability, resulting in a good PCE, can be used as efficient materials in non-fullerene OSCs.

### 2.4 Reorganization energy

The reorganization energy (RE) is an important feature to recognize the relationship between a compound structure and its capacity to transfer charge efficiently.<sup>33</sup> The performance and working capability of OSCs is mainly dependent upon the reorganization energy, which is actually the electron and hole



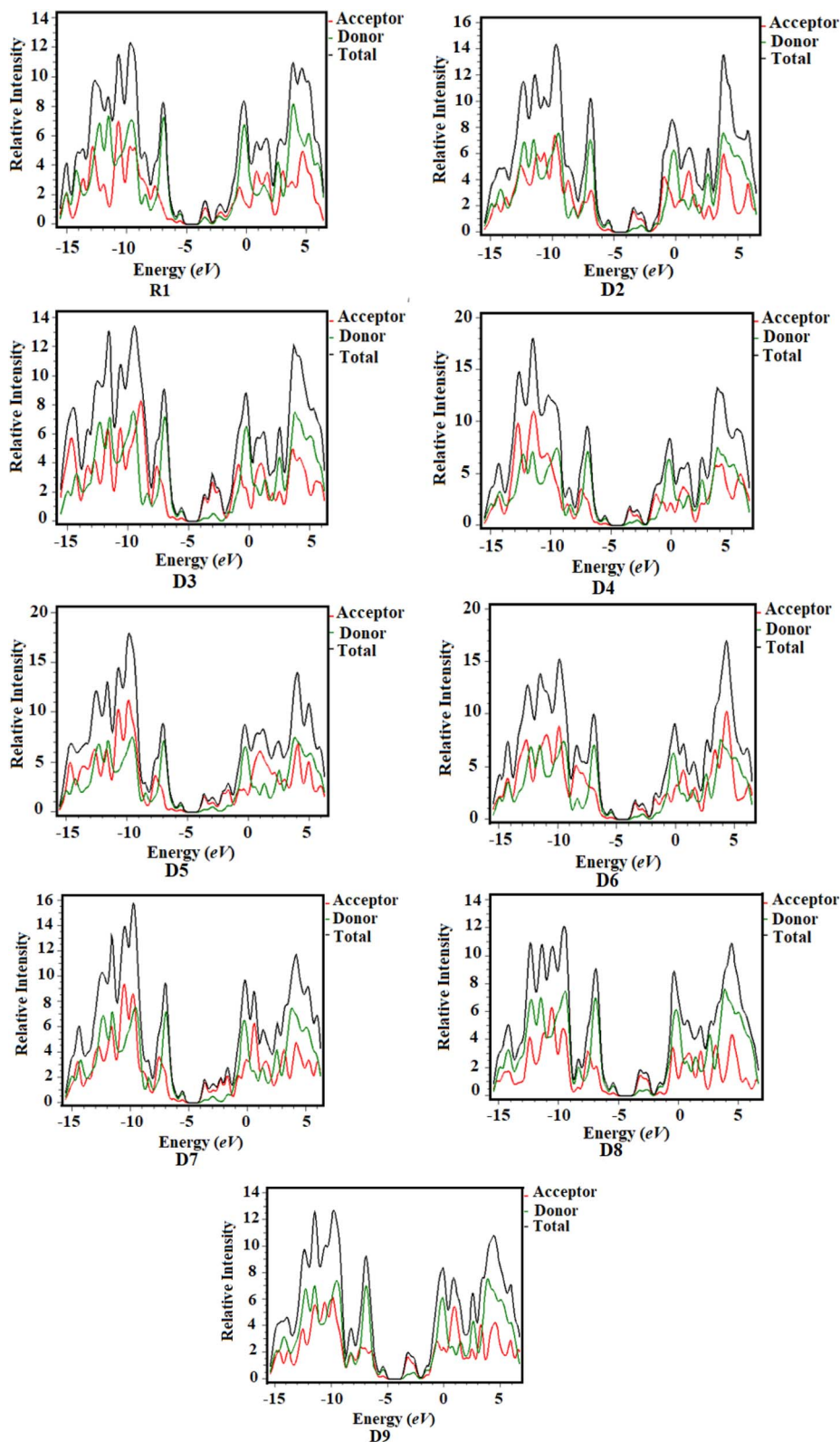


Fig. 5 Density of states (DOS) of the studied chloroforms as represented graphically.

transport ability of different materials.<sup>34</sup> Therefore, in order to understand the charge mobility, *i.e.*, the hole mobility ( $\lambda_h$ ) and electron mobility ( $\lambda_e$ ), of the studied chromophores, the RE was calculated at the B3LYP/6-31G(d,p) functional level (Table 4).

Essentially, the RE can be categorized into the internal reorganization energy ( $\lambda_{int}$ ) and external reorganization energy ( $\lambda_{ext}$ ).  $\lambda_{int}$  provides knowledge about quick changes in the internal structures, whereas  $\lambda_{ext}$  only deals with the external



Table 5 Open-circuit voltage of the studied compounds<sup>a</sup>

Compounds	$V_{oc}$ (V)	$\Delta E$
<b>R1</b>	1.30	2.003
<b>D2</b>	1.395	2.000
<b>D3</b>	1.131	1.851
<b>D4</b>	1.322	1.967
<b>D5</b>	1.182	1.891
<b>D6</b>	1.352	1.970
<b>D7</b>	1.186	1.892
<b>D8</b>	1.576	2.132
<b>D9</b>	1.574	2.103

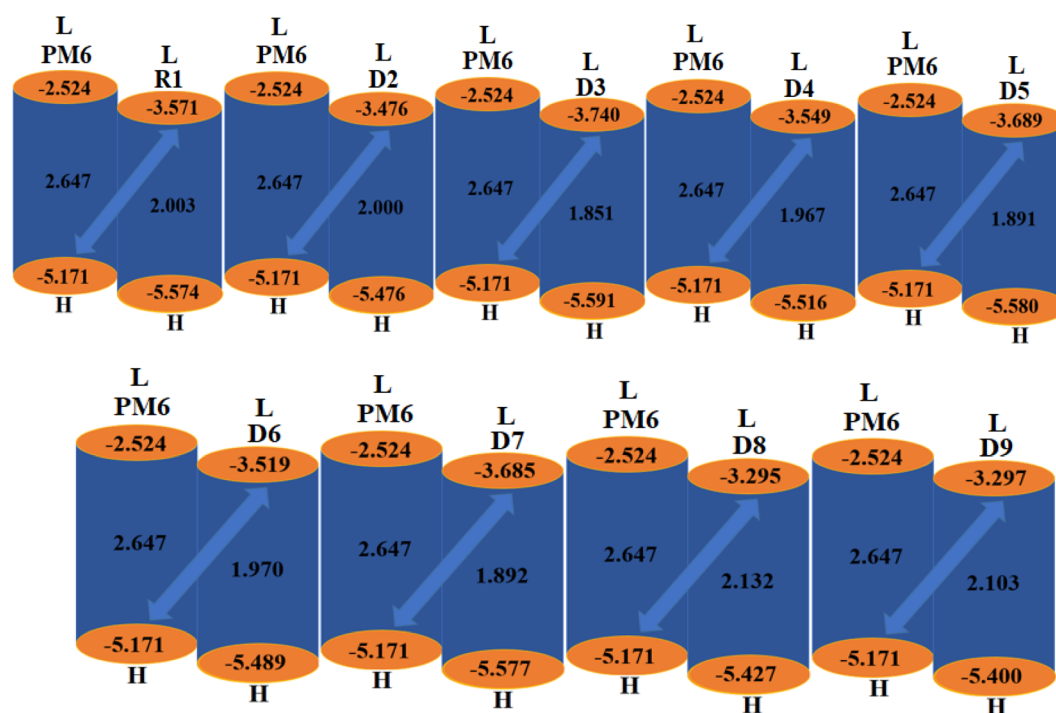
$$^a \Delta E = E_{LUMO}^A - E_{HOMO}^D$$

structural environment.  $\lambda_{ext}$  has little impact; hence this factor was overlooked in favor of the internal reorganizational energy. The computed  $\lambda_e$  of the reference and derivatives are found to be  $-0.00002673$ ,  $-0.00000762$ ,  $0.00003289$ ,  $-0.00003688$ ,  $0.00005186$ ,  $-0.00004563$ ,  $0.00002929$ ,  $0.00026435$  and  $0.0001786$  eV, respectively. **D6** has the lowest  $\lambda_e$  value ( $-0.00004563$  eV) of the studied molecules, indicating its faster electron mobility. The descending order of  $\lambda_e$  is **D6** > **D4** > **R1** > **D2** > **D7** > **D3** > **D5** > **D9** > **D8**. Consequently, for **R1** and **D2–D9**, the predicted  $\lambda_h$  values are  $0.00018773$ ,  $-0.33841872$ ,  $0.00009158$ ,  $0.00005475$ ,  $0.00007521$ ,  $0.00011363$ ,  $0.00008763$ ,  $0.00009166$ , and  $0.00009157$  eV, respectively. For the designated molecules, the increasing order of  $\lambda_h$  is **D2** > **D4** > **D5** > **D7** > **D9** > **D3** > **D8** > **D6** < **R1**. The overall discussion concludes that the lower  $\lambda_h$  values of all the derivatives expressed a greater rate of hole transport in them than in the reference molecule, except

for **D2**. This investigation reveals a greater charge-transport rate in the aforementioned chromophores, which indicated their potential as effective candidates for OSCs.

## 2.5 Density of states (DOS)

DOS analysis is considered to be an efficient tool to examine the charge densities on the chromophores and provides support for the investigations performed in the FMO study.<sup>35,36</sup> Herein, DOS analysis was executed for the studied chromophores to examine the charge-transfer rate, and the resulting graphs are shown in Fig. 5. We separated our compounds into two groups for the purpose of explaining the DOS analysis, *i.e.*, the central donor and sideways acceptors, as shown with black and red lines in the maps, respectively. The electronic distribution patterns on the acceptors for **R1** and **D2–D9** are found to be 23.9%, 23.1%, 23.7%, 23.1%, 23.6%, 23.1%, 23.6%, 22.0% and 22.8% for the HOMO and 66.5%, 79.4%, 83.8%, 79.8%, 81.7%, 80.8%, 82.3%, 70.4% and 75.4% for the LUMO, respectively. Similarly, the donor contributes 76.1%, 76.9%, 76.3%, 76.9%, 76.4%, 76.9%, 76.4%, 78.0% and 77.2% to the HOMO and 33.5%, 20.6%, 16.2%, 20.2%, 18.3%, 19.2%, 17.7%, 29.6% and 24.6% to the LUMO in **R1** and **D2–D9**, respectively.<sup>37</sup> For the HOMO, the charge density is predominantly present on the donor moiety but partly on the acceptor units, whereas the LUMO is found mostly on the acceptor components, as shown by the green and red peaks in the maps. DOS analysis revealed a high degree of charge coherence from the central donor to the end-group acceptor region, especially near the end-capped groups, as well as a significant delocalization of the electronic structure, as discussed in the FMO analysis.

Fig. 6 Graphical representation of the  $V_{oc}$  for the studied chromophores with PM6.



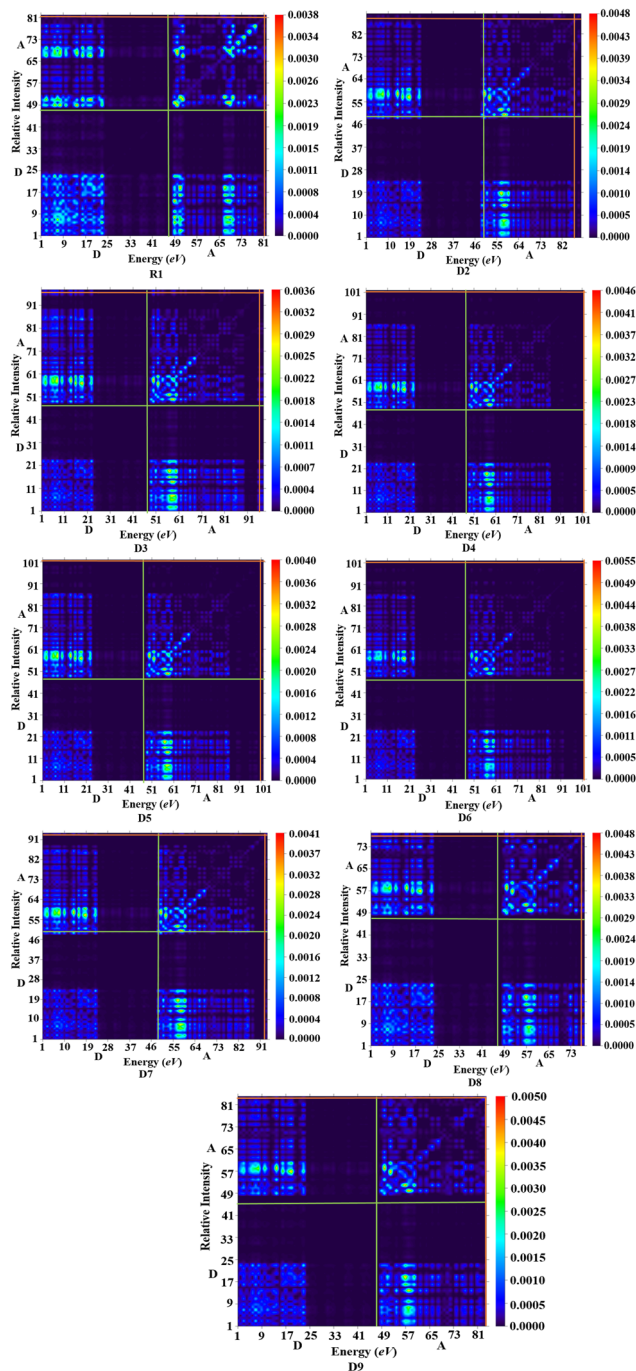


Fig. 7 TDM analysis in the reference state S1 and for the designed compounds.

## 2.6 Open-circuit voltage ( $V_{oc}$ ) analysis

The  $V_{oc}$  is an important parameter that plays a vital role in determining the working potential of OSCs<sup>38,39</sup> and is defined as the highest electric current that can readily be obtained from an optical material.<sup>40</sup> Certain key factors, such as light intensity, light source, charge-carrier recombination, external fluorescence proficiency and work functions of the electrodes, greatly affect the  $V_{oc}$ .<sup>41</sup> The  $V_{oc}$  can be calculated by taking the difference of the HOMO of the donor polymer molecule and the

Table 6 Calculated  $E_b$  of the R1 and D2–D9 compounds<sup>a</sup>

Compounds	$E_{H-L}$	$E_{opt}$	$E_b$
R1	2.003	1.724	0.279
D2	2.000	1.681	0.319
D3	1.851	1.561	0.290
D4	1.967	1.658	0.309
D5	1.891	1.593	0.298
D6	1.97	1.658	0.312
D7	1.892	1.592	0.300
D8	2.132	1.816	0.316
D9	2.103	1.770	0.333

<sup>a</sup> Units in eV.

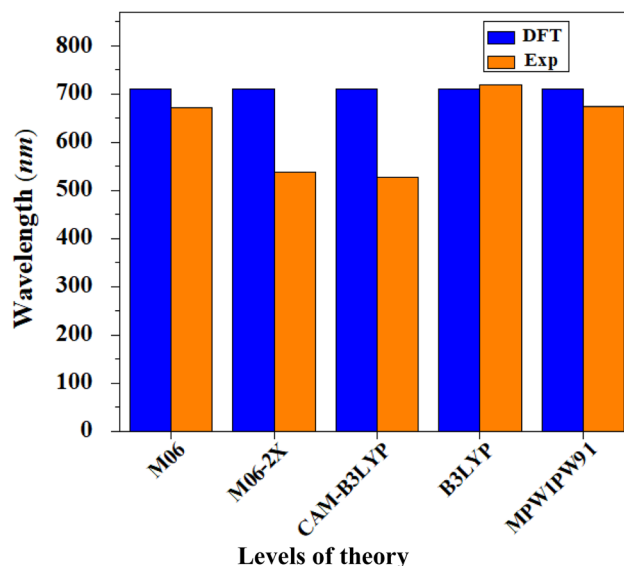


Fig. 8 Comparison between the theoretical and experimental UV-visible results of R1 in chloroform at various levels with 6-31G(d,p).

LUMO of the acceptor molecule. To achieve a higher  $V_{oc}$ , the LUMO of the acceptor moiety should be higher than the HOMO of the donor, as this reduces the band gap.<sup>42</sup> The theoretically computed results of  $V_{oc}$ , calculated using the equation given by Scharber *et al.*,<sup>43</sup> are presented in Table 5.

$$V_{oc} = (|E_{HOMO}^D| - |E_{LUMO}^A|) - 0.3 \quad (1)$$

Here,  $E$  denotes the elementary charge of the acceptors, signifying the charge on each molecule, and 0.3 is the empirical constant. A well-known donor polymer, PM6,<sup>44</sup> with HOMO/LUMO energies of 5.171 and  $-2.524$  eV, respectively, has been utilized to examine the  $V_{oc}$  values, as shown in Table 5.

The computed  $V_{oc}$  value of the reference chromophore shows excellent agreement with its experimental value of 0.86 V,<sup>19</sup> which indicated the suitable selection of the level of theory for our study of the photovoltaic properties of R1 and D2–D9, whose LUMO<sub>acceptor</sub> values are 1.3, 1.395, 1.131, 1.322, 1.182, 1.352, 1.186, 1.576 and 1.574 V, respectively. The  $V_{oc}$  of the studied chromophores decreased in the order of D8 > D9 > D2 >



**D6 > D4 > R1 > D7 > D5 > D3**. Among all the compounds, **D8** had the highest  $V_{oc}$  value (1.576 V). The other designed derivatives showed comparable open-circuit voltages with that of the reference molecule. A low-lying LUMO leads to improved  $V_{oc}$  values with an enhanced photovoltaic response. The orbital energy diagrams of all the investigated chromophores blended with **PM6** are shown in Fig. 6. It is clear from Fig. 6 that the LUMO level of the acceptor molecules (**R1** and **D2–D9**) lies below the HOMO of the donor **PM6** polymer. This kind of placement of molecular orbitals enables facile transfer of the electron cloud from **PM6** to the acceptor, leading to enhanced optoelectronic behavior.

### 2.7 Transition density matrix (TDM)

Various types of transitions in the reference and tailored chromophores were evaluated *via* TDM analysis. TDM study elucidates the charge transport inside molecules, making it a crucial analysis for determining the performance of OSCs. Applying the B3LYP/6-31G(d,p) functional, the emission and absorption of electronic charges for S1 from the S0 state were calculated (Fig. 7). The contribution of hydrogen to the transition is so minor that it is neglected in the current study. The TDM approach is crucial for estimating the excited state phenomena of electronic excitation and electron–hole localization.<sup>45,46</sup> **R1** and **D2–D9** are split into two segments: the acceptor (the end-capped molecule: A) and donor (the core component: D). Fig. 7 indicates that effective charge is present at the terminal acceptor moieties, as higher-electron intensity green and red spots can be seen over those portions. According to the results obtained from the TDM heat maps, it has been observed that the electronic charge is efficiently transferred in a diagonal way from the donor to the acceptor in all the derivatives without trapping, thus showing charge coherence.

### 2.8 Exciton binding energy ( $E_b$ )

$E_b$  is defined as the difference between the HOMO and LUMO with the first singlet-to-singlet excitation state energy ( $E_{opt}$ ) and is an important component that aids researchers in exploring the photovoltaic response of OSCs.<sup>47</sup> The interaction of the coulombic forces between the electrons ( $\lambda_e$ ) and the holes ( $\lambda_h$ ) is measured by the binding energy. Low-binding-energy molecules have a high power conversion efficiency and a high current charge density ( $J_{sc}$ ).<sup>48,49</sup> Eqn (2) was used to compute the  $E_b$  of the **R1** and **D2–D9** acceptor molecules, where  $E_{H-L}$  represents the energy difference between the HOMO and LUMO.

$$E_b = E_{H-L} - E_{opt} \quad (2)$$

Table 6 shows the DFT-estimated findings of the  $E_b$  of the studied compounds in chloroform.

Table 6 shows that all the derivatives exhibited comparable values of binding energies with their reference chromophores. For all the molecules, the descending pattern of  $E_b$  is as follows: **D9 < D2 < D8 < D6 < D4 < D7 < D5 < D3 < R1**. Structures with an  $E_b$  of 2.1 eV or below are generally considered to be efficient OSC materials with a high  $V_{oc}$ .<sup>49</sup> Interestingly, all our compounds

exhibited  $E_b$  values lower than 2.1 eV, which illustrates their high exciton dissociation rate and large charge transfer rate, as shown in the TDM maps. All these investigations suggest that these designed molecules would be effective photovoltaic materials.

## 3 Conclusions

Advanced quantum chemistry techniques have been successfully employed to investigate the optoelectronic, photophysical and photovoltaic properties of newly designed A–D–A-type fullerene-free chromophores. Terminal structural tailoring has proven to be the most important strategy for developing high-performance photovoltaic materials with good optoelectronic properties for efficient OSCs. The FMO findings reveal a diminishing energy gap between the HOMO and LUMO with effective charge transfer in the derivatives, which is further supported by TDM and DOS investigations. Comparing the absorption properties, the derivatives show a red shift in their emission spectra ( $\lambda_{max} = 794.516$ – $737.562$  nm in chloroform and  $744.784$ – $693.928$  nm in the gas phase) compared to that of **R1** ( $\lambda_{max} = 719.082$  nm in chloroform and  $673.242$  nm in the gas phase), except for **D8** and **D9**. Furthermore, TDM analysis demonstrated that the end-capped substituents could rapidly extract electron density, resulting in an increased charge-transfer rate when tailored with the acceptors. Moreover, the  $V_{oc}$  was also estimated with regards to HOMO<sub>PM6</sub>–LUMO<sub>acceptor</sub> and all the derivatives show comparable voltage results with the reference molecule. Our predicted molecules might have good photovoltaic and electronic properties, as evidenced by their excitation binding energies and moderate reorganization. According to our findings, end-capped terminal modification is a useful technique for developing novel and highly competitive photovoltaic materials with good electrical and optical properties. It is suggested that the predicted compounds should be synthesized to create highly efficient organic solar cells.

## 4 Computational procedure

All the computations of this study were performed with the aid of the Gaussian 09 (ref. 50) program. Gauss View 5.0 (ref. 51) was used to create the input data and to display the results. To optimize the geometry of the reference compound (**R1**) without symmetry reduction, five exchange–correlation (XC) functionals were utilized: B3LYP,<sup>52</sup> CAM-B3LYP,<sup>53</sup> MPW1PW9,<sup>54</sup> M06 (ref. 55) and M06-2X,<sup>56</sup> with the 6-31G(d,p) basis set combination. To select a good theoretical technique, the computed maximum absorbed wavelength of the reference chromophore **R1** in chloroform with the abovementioned functionals was compared to the experimentally reported data. The maximum values of the reference compound generated employing the aforementioned functionals were 719.082, 674.487, 526.45, 537.799 and 671.965 nm, respectively, whereas the maximum value of **ITBCR** measured experimentally was 711 nm.<sup>19</sup> At the B3LYP/6-31G(d,p) level of DFT, a good agreement with the experimental findings was seen (Fig. 8); therefore, this functional was selected for further investigations. Moreover, to

visualize the optoelectronic and photovoltaic properties, various analyses, such as FMO, DOS, TDM, binding energy, and open-circuit voltage analysis, were performed for **R1** and **D2–D9**. The internal ( $\lambda_{\text{int}}$ ) and external ( $\lambda_{\text{ext}}$ ) reorganization energies are the two fundamental types of reorganization energy. The first one,  $\lambda_{\text{int}}$ , is based on internal structural modification, whereas  $\lambda_{\text{ext}}$  deals with the influence of polarization in the external environment.<sup>57</sup> In this report, factors related to external environmental changes were not considered and our only focus was on  $\lambda_{\text{int}}$ .<sup>25</sup> The energies for hole ( $\lambda_{\text{h}}$ ) and electron ( $\lambda_{\text{e}}$ ) rearrangement were calculated using the equations below.<sup>58</sup>

$$\lambda_{\text{e}} = [E_0^- - E_-] + [E_0^0 - E_0] \quad (3)$$

$$\lambda_{\text{h}} = [E_0^+ - E_+] + [E_0^0 - E_0] \quad (4)$$

where  $E_+^0$  and  $E_-^0$  represent the neutral compound's capabilities as a result of the optimized cation and anion geometries;  $E_-$  and  $E_+$  indicate the energies of anions and cations, respectively;  $E_0^-$  and  $E_0^+$  represent the cations and anions exhibiting different single point energies (SPEs); and  $E_0$  is the neutral molecule of SPE.<sup>59</sup>

## Conflicts of interest

No conflicts to declare.

## Acknowledgements

The authors are grateful to the cooperation and collaboration of A. A. C. B. from IQ-USP Brazil, especially for his continuous support and providing computational lab facilities. Dr Muhammad Khalid gratefully acknowledges the financial support of HEC Pakistan (project no. 20-14703/NRPU/R&D/HEC/2021). M. A. A. appreciates the support of the Research Center for Advanced Materials Science (RCAMS) at King Khalid University Abha, Saudi Arabia, through grant KKU/RCAMS/G015/21.

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