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## Fullerene C<sub>70</sub>/porphyrin hybrid nanoarchitectures: single-cocrystal nanoribbons with ambipolar charge transport properties†

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In recent years, supramolecular cocrystals containing organic donors and acceptors have been explored as active components in organic field-effect transistors (FETs). Herein, we report the synthesis of novel single-cocrystal nanoribbons with ambipolar charge transport characteristics from C<sub>70</sub> and 5,10,15,20-tetrakis(3,5-dimethoxyphenyl)porphyrin (3,5-TPP) in a 3 : 2 ratio. The C<sub>70</sub>/3,5-TPP nanoribbons exhibited a new strong absorption band in the near-infrared region, indicating the presence of charge-transfer interactions between C<sub>70</sub> and 3,5-TPP in the cocrystals. We elucidated the mechanism of the charge-transport properties of the nanoribbons using photoemission yield spectroscopy in air and theoretical calculations. A strong interaction between porphyrins in the one-dimensional porphyrin chains formed in C<sub>70</sub>/3,5-TPP nanoribbons, which was confirmed by single-crystal X-ray diffraction, plays a crucial role in their hole transport properties.

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### Introduction

Functional supramolecular nanoarchitectures have tremendous potential as building blocks for bottom-up organic nanoscale electronic devices<sup>1–3</sup> such as transistors.<sup>4</sup> Recently, several studies have been reported on transistors employing supramolecular nanoarchitectures, such as nanowiskers, nanotubes, nanorods, nanowires, and nanosheets.<sup>1,5,6</sup> A distinct characteristic of almost all these reported transistors is that they transport only a single carrier type, *i.e.*, either holes (p-type) or electrons (n-type). Therefore, simultaneous or selectable transport of electrons and/or holes (ambipolar charge

transport) remains a highly desirable device characteristic to be achieved, because this characteristic will be able to facilitate the design of better-performing electronic circuits<sup>7</sup> as well as the demonstration of bifunctional organic devices, such as light-emitting<sup>8</sup> and light-sensing transistors.<sup>9</sup> In 2012, we reported novel ambipolar FETs based on C<sub>60</sub>/5,10,15,20-tetrakis(4-methoxyphenyl)porphyrinato cobalt(II) (C<sub>60</sub>/CoTPP) cocrystals.<sup>10</sup> Zhu *et al.* also reported ambipolar FETs using sulfur-bridged annulene-TCNQ cocrystals.<sup>11</sup> Following these earlier studies, various FETs based on donor-acceptor (D-A) cocrystals have been reported.<sup>12–14</sup> There are several reports on ambipolar FETs based on novel organic semiconductors in the literature,<sup>15,16</sup> however, the preparation of these semiconductors requires complex organic syntheses. In contrast, the co-crystallization process can be conducted by simple mixing of two (or more) molecules and does not involve any complex synthesis protocols. Notably, not all D-A cocrystal-based FETs exhibit ambipolar transport properties, some exhibit the unipolar ones, such as n-type<sup>17–20</sup> or p-type.<sup>21</sup> Therefore, further research is necessary to understand these characteristics in detail.

Herein, we report the preparation of cocrystal nanoribbons from C<sub>70</sub> and 5,10,15,20-tetrakis(3,5-dimethoxyphenyl)porphyrin (3,5-TPP) as a pair of electron-acceptor and electron-donor molecules by a simple solution mixing method. These C<sub>70</sub>/3,5-TPP nanoribbons were used to fabricate bottom-gate/bottom-contact FETs. Despite using a non-optimized device architecture, these FETs exhibited ambipolar transport

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characteristics. This result is in sharp contrast with our earlier report on  $C_{60}$  analogue ( $C_{60}/3,5\text{-TPP}$  cocrystals) that showed only n-type behavior.<sup>17</sup> We found that the one-dimensional porphyrin chains formed by the strong interaction of porphyrins inside the nanoribbons play a very important role in the hole transport properties of the  $C_{70}/3,5\text{-TPP}$  nanoribbons.

## Experimental section

### Preparation and characterization of $C_{70}/3,5\text{-TPP}$ nanoribbons

$C_{70}/3,5\text{-TPP}$  nanoribbons were prepared by mixing two solutions containing  $C_{70}$  and 3,5-TPP. Specifically, 3 ml of  $C_{70}$ -saturated toluene solution was placed in a 9 ml glass bottle and 1 mg of 3,5-TPP in 1 ml of toluene was added to this solution. The mixed solution was then vigorously shaken and stored at 10 °C for 24 h to grow the  $C_{70}/3,5\text{-TPP}$  nanoribbons.  $C_{70}/3,5\text{-TPP}$  nanoribbons were collected by filtering the solution and washing with iso-propyl alcohol (IPA). These nanoribbons were stable in dark conditions at room temperature for long periods.

The structure and morphology of the obtained nanoribbons were analyzed by scanning electron microscopy (SEM; JEOL JSM-6700F, 5 kV), atomic force microscopy (AFM, JEOL, JSPM-5200), a micro-Raman system (Photon Design) equipped with a semiconductor laser with a wavelength of 532 nm, and a single crystal diffractometer (Rigaku Saturn CCD) using Mo  $K\alpha$  X-ray radiation at 113 K. The diffuse reflectance spectra of the nanoribbons were recorded using a UV-Vis-NIR spectrometer (JASCO V-570) equipped with an integrating sphere. Photo-emission yield spectroscopy in air (PYSA) was performed using a RIKEN KEIKI AC-3 photoelectron spectrometer equipped with a monochromatic D2 lamp.

The time-resolved luminescence (TRPL) spectra of materials were measured by excitation with a pulsed laser (100 fs pulse width, 1 kHz repetition frequency) generated by an optical parametric amplifier (Spectra Physics TOPAS Prime). The excitation wavelength for TRPL spectroscopy analysis was 520 nm.

Transistors were fabricated using a previously reported method.<sup>17</sup> The electrical transport properties were measured inside a glove box using an Agilent B2902A and Agilent E5272A.

For the first-principles quantum chemical calculations, the Gaussian16 (Revision A.03) program package<sup>22</sup> was employed together with our Python-based quantum chemical tool.<sup>23</sup> The electronic structures were obtained by using the density functional theory (DFT) at the B3LYP/6-31G(d) level. The molecular orbitals were depicted by using Chemcraft.<sup>24</sup>

## Results and discussion

The nanoribbons are classified as cocrystals of  $C_{70}$  and 3,5-TPP as they are single-phase crystalline materials, composed of two different molecules. Fig. 1 shows optical microscopy and SEM images of the  $C_{70}/3,5\text{-TPP}$  nanoribbons. Both the images show parallelogram morphology of the nanoribbons. The length and width of the nanoribbons were found to be 15–20  $\mu\text{m}$  and 0.2–0.5  $\mu\text{m}$ , respectively (Fig. S1†). The thickness of the nanoribbons, as determined by SEM and AFM measurements, was in the range of 20–200 nm (Fig. S2†). Fig. 2 shows the diffuse

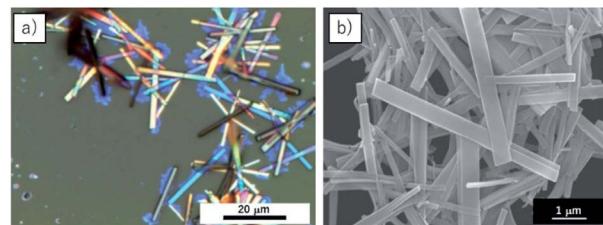


Fig. 1 (a) Optical microscopy image and (b) SEM image of  $C_{70}/3,5\text{-TPP}$  nanoribbons.

reflectance spectra of the  $C_{70}/3,5\text{-TPP}$  nanoribbons,  $C_{70}$  crystals, and 3,5-TPP film. It is observed that the  $C_{70}$  crystals prepared using the liquid–liquid interface precipitation method<sup>17</sup> showed a broad absorption in the 700–400 nm region, whereas the 3,5-TPP film exhibited strong and weak absorption at approximately 400 nm and 500–600 nm, respectively. The  $C_{70}/3,5\text{-TPP}$  cocrystals exhibited a new absorption in the longer wavelength region (up to 900 nm). Subtraction of the normalized absorption spectrum of  $C_{70}$  from that of the  $C_{70}/3,5\text{-TPP}$  cocrystals clearly showed the extra absorption in the range of 900–700 nm (with an absorption maximum at 760 nm), which was referred to as the charge-transfer (CT) absorption band of the  $C_{70}/3,5\text{-TPP}$  nanoribbons.

To understand the CT characteristics under light illumination, we measured the photoluminescence (PL) spectra of the  $C_{70}/3,5\text{-TPP}$  cocrystals (Fig. 3). The CT fluorescence band was observed in the range of 750–1000 nm with a peak at 800 nm, as a mirror image of the CT absorption band at 760 nm. These CT emission peaks are apparently in longer wavelength regions than the emission of  $C_{70}$ -crystals.<sup>25</sup> A similar CT band was reported for  $C_{60}/3,5\text{-TPP}$  cocrystals.<sup>17</sup> Based on the appearance of the CT absorption band, we conclude that  $C_{70}$  molecules interact with the nearest 3,5-TPP molecules. These interactions in the ground state lead to the formation of the  $C_{70}/3,5\text{-TPP}$  cocrystals with alternative arrangement of  $C_{70}$  and 3,5-TPP.

The fluorescence lifetime of the  $C_{70}/3,5\text{-TPP}$  cocrystals was evaluated from the decay-time profile measured using time-

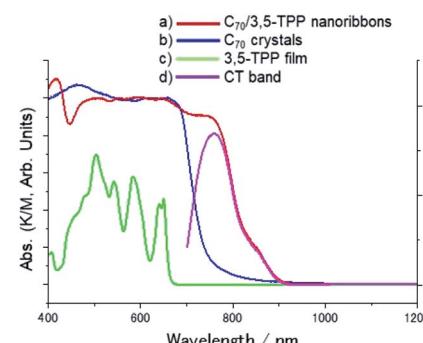


Fig. 2 Diffuse reflectance spectra in Vis and near-infrared regions (K/M: Kubelka–Munk function, which is proportional to absorbance); (a)  $C_{70}/3,5\text{-TPP}$  nanoribbons, (b)  $C_{70}$  crystals, (c) 3,5-TPP film, (d) subtraction of normalized absorption spectra at 610 nm as shown in (a) and (b).



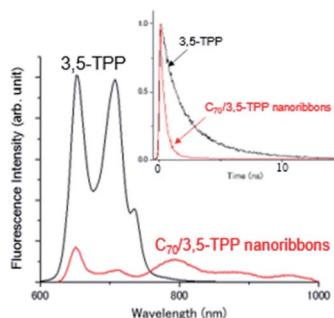


Fig. 3 Steady-state fluorescence spectra of 3,5-TPP crystals (black) and C<sub>70</sub>/3,5-TPP nanoribbons (red);  $\lambda_{\text{ex}}$ : 520 nm. Inset: Time profiles of fluorescence intensity [ $\lambda_{\text{ex}}$ : 520 nm,  $\lambda_{\text{em}}$ : 690–730 nm for 3,5-TPP crystals (black)<sup>17</sup> and  $\lambda_{\text{em}}$ : 780–820 nm for C<sub>70</sub>/3,5-TPP nanoribbons (red)].

resolved luminescence (TRPL) spectroscopy (inset of Fig. 3). Based on the luminescence decay at 770–810 nm, the lifetime of the CT luminescence was measured as approximately 0.5 ns, which is shorter than those of the 3,5-TPP crystal (approximately 2.5 ns) and C<sub>60</sub>/3,5-TPP cocrystals (approximately 1.0 ns), which is calculated using the previously reported data.<sup>17</sup> The observed CT luminescence lifetime suggests that the short CT-excited lifetime is due to rapid shift in electron distribution from the donor to the acceptor part in the excited CT state.

To investigate the charge transport properties of the C<sub>70</sub>/3,5-TPP nanoribbons, we fabricated bottom-gate, bottom-contact FETs by drop-casting an IPA solution containing C<sub>70</sub>/3,5-TPP nanoribbons onto a substrate. The substrate used was pre-patterned with gold source-drain electrodes with a channel length and width of 2–10  $\mu\text{m}$  and 10 000  $\mu\text{m}$ , respectively, and SiO<sub>2</sub> (300 nm thick) as the gate dielectric. The Au electrodes were treated with self-assembled monolayers of undecanethiol and the SiO<sub>2</sub> dielectric interface was rendered hydrophobic by treatment with hexamethyldisilazane. This fabrication method, as developed by Samori *et al.*,<sup>26</sup> is simple and convenient for the qualitative charge-transport characterization of the micrometer-

sized single-crystals. Fig. 4 shows the dependence of the drain current ( $I_{\text{D}}$ ) on the gate voltage ( $V_{\text{G}}$ ) of the C<sub>70</sub>/3,5-TPP nanoribbon FET. The measurements were performed under N<sub>2</sub> atmosphere at room temperature under dark conditions after annealing at 353 K. (By annealing at 353 K, we observed the increase of channel current of the devices. The XRD pattern measured after annealing showed that the crystal structure is unchanged even after annealing (Fig. S3†).) The transfer characteristics of the C<sub>70</sub>/3,5-TPP nanoribbon-based FETs showed a V-shape in which the two arms indicated electron transport (n-type) and hole transport (p-type), respectively. These results are in sharp contrast with previously reported measurements on C<sub>60</sub> nanowhisker<sup>5</sup>- and C<sub>60</sub>/3,5-TPP cocrystal<sup>17</sup>-based FETs that showed only n-type behavior. The electron and hole mobilities calculated from the data in Fig. 4 were relatively low, in the order of  $8.2 \times 10^{-5} \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$  and  $3.1 \times 10^{-6} \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$ , respectively. However, we believe that further device optimization can lead to higher carrier mobilities.

There are many studies in the literature on D–A cocrystal-based FETs reporting n-type, p-type, or ambipolar characteristics;<sup>17–21</sup> however, their characteristics and charge transport mechanisms have not yet been fully understood. In order to understand the mechanism of the ambipolar charge transport properties, a comparison of C<sub>70</sub>/3,5-TPP cocrystals observed in the present study with our earlier report on C<sub>60</sub>/3,5-TPP cocrystals<sup>17</sup> that showed only n-type behavior may be useful.

At first, to realize their electronic structure, we carried out PYSA measurements of C<sub>70</sub>/3,5-TPP and C<sub>60</sub>/3,5-TPP cocrystals, 3,5-TPP, and C<sub>70</sub> powder. PYSA is a powerful tool for understanding the electronic and electrical properties of molecular semiconductors.<sup>27–29</sup> Recently, we succeeded in evaluating the

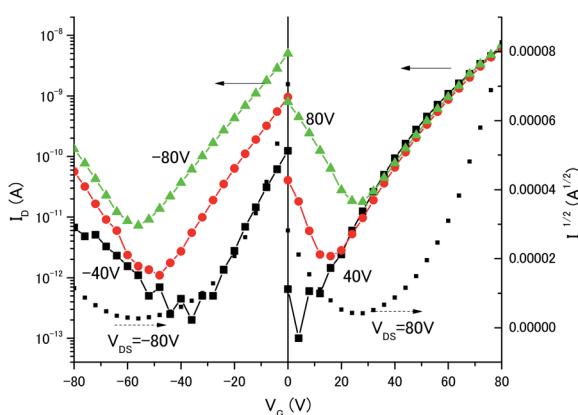


Fig. 4 Transfer ( $I_{\text{D}}$ – $V_{\text{G}}$ ) characteristics of C<sub>70</sub>/3,5-TPP nanoribbons recorded in dark conditions. The solid curve displays drain current ( $I_{\text{D}}$ ) as a function of gate voltage ( $V_{\text{G}}$ ), and the dotted curve shows the square root of  $I_{\text{D}}$  (right vertical axis).

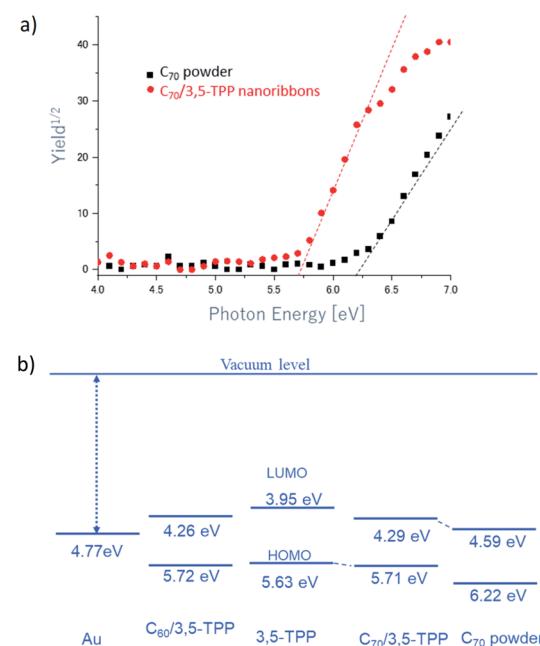


Fig. 5 (a) PYSA spectra and (b) energy level diagrams of C<sub>60</sub>/3,5-TPP cocrystals, 3,5-TPP, C<sub>70</sub>/3,5-TPP nanoribbons, and C<sub>70</sub> powder. For semi-conductor, the HOMO and the LUMO correspond to the top of the valence band and bottom of the conduction band, respectively.



electronic structure of  $C_{60}$ /Fc nanosheets with high sensitivities using PYSA.<sup>30</sup>

The ionization energy ( $I_s$ ) of the  $C_{70}$ /3,5-TPP nanoribbons was determined to be 5.71 eV, which is 0.51 eV smaller than that of the  $C_{70}$  powder, similar to that of 3,5-TPP. Fig. 5b shows the energy level diagrams of the  $C_{70}$ /3,5-TPP nanoribbons,  $C_{60}$ /3,5-TPP cocrystals, 3,5-TPP, and  $C_{70}$ , estimated using the measured values of  $I_s$  and the energy gap ( $E_g$ ). The  $E_g$  values were obtained from the Vis-NIR absorption spectra measured in the present study for the  $C_{70}$ /3,5-TPP nanoribbons and the previous study for  $C_{60}$ /3,5-TPP cocrystals.<sup>17</sup> The energy level diagrams demonstrate that the cocrystallization of  $C_{70}$  with 3,5-TPP enhanced the p-type FET characteristics because of the decreased hole injection barrier between the gold electrode and the highest occupied molecular orbital (HOMO) of the  $C_{70}$ /3,5-TPP nanoribbons. Similarly, the  $I_s$  of the  $C_{60}$ /3,5-TPP cocrystals was determined to be 5.72 eV (Fig. S4†), and the calculated energy level of the lowest unoccupied molecular orbital (LUMO) of the  $C_{60}$ /3,5-TPP cocrystals is located close to that of  $C_{70}$ /3,5-TPP nanoribbons. This indicates that the decrease in the hole injection barrier also takes place in the  $C_{60}$ /3,5-TPP cocrystal-FET. Therefore the difference between the charge transport properties of ambipolar  $C_{70}$ /3,5-TPP nanoribbons and unipolar (n-type)  $C_{60}$ /3,5-TPP cocrystals cannot be explained only by the decrease in the hole injection barrier. Consequently, some other factors such as structural difference must be considered.

To determine the relationship between the charge transport properties and the molecular arrangement in the  $C_{70}$ /3,5-TPP nanoribbon, we determined the crystal structure of the  $C_{70}$ /3,5-TPP nanoribbon by single-crystal XRD at 113 K (Fig. 6a). Crystal structure of the  $C_{70}$ /3,5-TPP nanoribbon belongs to triclinic  $P\bar{1}$  space group with the unit cell:  $a = 13.8828(3)$  Å,  $b = 20.9103(5)$  Å,  $c = 30.2321(7)$  Å,  $\alpha = 84.678(1)$ °,  $\beta = 86.015(1)$ °, and  $\gamma = 84.537(1)$ ° (CCDC-2161616) (Fig. 6). The cocrystal was formulated as  $[(3,5\text{-TPP})_2 \cdot (C_{70})_3 \cdot (\text{toluene})_3]$  and contained a  $[C_{70}/(3,5\text{-TPP})/C_{70}]_2$  unit, in which four  $C_{70}$  molecules assembled above and below the two 3,5-TPP molecular planes (Fig. 6c). The  $C_{70}$  molecules were found to be arranged in zigzag chains that render effective electron transport from one  $C_{70}$  to a nearby  $C_{70}$ . This type of  $[(\text{fullerene}/(3,5\text{-TPP})/\text{fullerene})]$  substructure was also observed in the  $C_{60}$ /3,5-TPP cocrystal.<sup>17</sup> One of the biggest structural differences between the  $C_{70}$ /3,5-TPP nanoribbons and  $C_{60}$ /3,5-TPP cocrystals is the distance between the nearest 3,5-TPP molecules in the crystal structure. In the  $C_{70}$ /3,5-TPP nanoribbons, the 3,5-TPPs were found to be located very close to each other and interacted with the 3,5-dimethoxyphenyl group, resulting in the formation of one-dimensional porphyrin chains (Fig. 6d). The center-to-center distance between the two 3,5-TPP was measured to be approximately 15 Å. Based on the observed results, the most probable mode for electric conduction stems from these porphyrin chains which act as hole-transport pathways in the  $C_{70}$ /3,5-TPP nanoribbons by the hopping conduction. One-dimensional porphyrin chains are also formed in  $C_{60}$ /3,5-TPP cocrystals. However, the distance between the two 3,5-TPP is approximately 21 Å, because two toluene molecules exist between two 3,5-TPP molecules, which reduces the interaction of the 3,5-TPPs (Fig. S5†). As a result, the  $C_{60}$ /3,5-TPP co-crystal

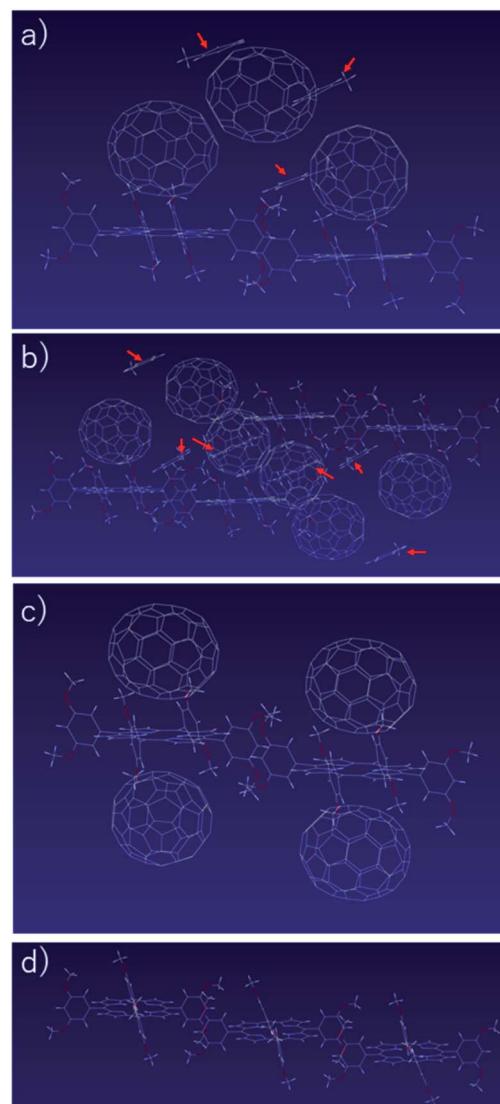


Fig. 6 Crystal structure of a  $C_{70}$ /3,5-TPP nanoribbons, (a)  $[(3,5\text{-TPP})_2 \cdot (C_{70})_3 \cdot (\text{toluene})_3]$  unit structure (b)  $[(3,5\text{-TPP})_2 \cdot (C_{70})_3 \cdot (\text{toluene})_3]_2$  unit structure (c) the  $[C_{70}/(3,5\text{-TPP})/C_{70}]_2$  unit structure, and (d) one-dimensional porphyrin chains unit structure. Red arrows indicate the toluene molecules in the crystals.

exhibited only n-type behavior. Whereas, in the  $C_{70}$ /3,5-TPP nanoribbons, toluene molecules are located above the 3,5-TPP molecules (Fig. 6a and b and S6†). These geometrical differences may be related to the fact that ambipolar transport characteristics were observed only for the  $C_{70}$ /3,5-TPP nanoribbons, but not for the  $C_{60}$ /3,5-TPP cocrystals.

For better understanding the hole transport mechanism of cocrystals, we estimated the coupling constant between 3,5-TPPs, based on the DFT method. Fig. 7a and b show intermolecular structures extracted from  $C_{60}$ - and  $C_{70}$ -based cocrystals with 3,5-TPP, respectively, where TPPs of the  $C_{70}$ -based cocrystal are more densely packed, and the distance between side chains of porphyrin derivatives is shorter, compared with those of  $C_{60}$ /3,5-TPP. The coupling constant rapidly decreases with inter-molecular distances, and therefore hole can more



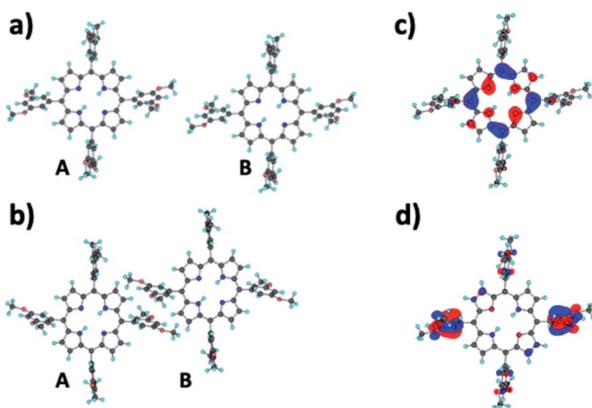


Fig. 7 TPPs structures extracted (a)  $C_{60}$ /3,5-TPP and (b)  $C_{70}$ /3,5-TPP cocrystals. The distance between porphyrins in  $C_{70}$ /3,5-TPP is shorter than that in  $C_{60}$ /3,5-TPP. The molecular orbital of (c) HOMO and (d) HOMO-3, which are localized at the center and side chains of TPP, respectively. The interaction between side chains of 3,5-TPPs gives a larger coupling constant.

easily transport in the  $C_{70}$ /3,5-TPP nanoribbons. The first-principles quantum chemical method gives the coupling constant of  $4.1 \times 10^{-4}$  and  $5.3 \times 10^{-2}$  eV between TPPs of  $C_{60}$ - and  $C_{70}$ -based cocrystals, respectively. Here, the coupling constant is estimated from  $\langle \phi_A | H_{AB} | \phi_B \rangle$ , where  $\phi_{A(B)}$  is the molecular orbital of each TPP molecule, and  $H_{AB}$  is the Hamiltonian of the whole molecular system. We note that there are several techniques to estimate coupling constants. For example, Krysko *et al.* employed the multireference quantum chemistry method.<sup>31</sup> The generalized Mulliken–Hush method was adopted to calculate electronic couplings for hole transfers in DNA.<sup>32</sup> In the coupling constant calculations for the hopping process, the localization of hole may be a key factor. This paper assumes that a hole is localized on each 3,5-TPP site. Fig. 7c and d show that the molecular orbitals of HOMO and HOMO-3 are localized at the center and side-chains of 3,5-TPP, respectively. The coupling constant has a larger value between molecular orbitals localized on side chain, such as seen in Fig. 7d. However, the coupling constant between HOMOs is small ( $8.8 \times 10^{-4}$  eV) even in the  $C_{70}$ /3,5-TPP case. We also note that the coupling constant related to HOMO-2 becomes relatively large because of localized orbitals on side chains. The details on molecular orbitals and coupling constants are summarized in the ESI (Fig. S7 and Table S1†). The molecular design on the side chain may be essential to control fast hole transports. Conversely, the large distance between side chains of  $C_{60}$ /3,5-TPPs disturbs efficient hole transfers, and results in the n-type behavior of the cocrystal. Thus, the hole conductivity of cocrystal materials is also controlled by higher-order structures such as nanoribbons.

## Conclusions

We successfully constructed supramolecular nanoarchitectures (nanoribbons) comprising  $C_{70}$  and 3,5-TPP in a 3 : 2 ratio using a simple solution mixing method. The  $C_{70}$ /3,5-TPP nanoribbons

exhibited ambipolar transport characteristics with nearly balanced hole/electron mobilities. This result is in sharp contrast with our earlier report on  $C_{60}$ /3,5-TPP cocrystals, which showed only n-type behavior. Single-crystal X-ray diffraction showed that one-dimensional porphyrin chains with zigzag  $C_{70}$  arrangement are formed in  $C_{70}$ /3,5-TPP cocrystals. Our theoretical calculations show the one-dimensional porphyrin chains formed by the strong interactions through the side chains of porphyrins play a very important role in the hole transport properties of  $C_{70}$ /3,5-TPP nanoribbons, in addition to the low electron injection barrier. Meanwhile, in the case of  $C_{60}$ /3,5-TPP cocrystals, the interaction between porphyrins is reduced owing to the presence of two toluene molecules between porphyrins, which hinder hole transport from one porphyrin to a nearby porphyrin in the  $C_{60}$ /3,5-TPP cocrystals. These observations are also supported by the theoretical calculations.

In recent years, numerous organic semiconductors with narrow bandgaps have been synthesized to fabricate ambipolar FETs. In the present study, we developed a distinctive method to prepare ambipolar materials without synthesizing new narrow-bandgap organic semiconductors. The successful preparation of nanoarchitectures with unique electronic characteristics by a simple co-crystallization method can be considered as an important stepping stone toward the fabrication of novel nanoscale devices.

## Author contributions

K. N. and C. H. fabricated and characterized the devices. K. i. N., K. F., and O. I. contributed to the interpretation of the experimental results. Y. M. performed single-crystal XRD analysis. Y. W. measured TRPL spectra. M. T. and T. S. performed theoretical calculations. S. Y., Y. L., and Y. N. measured the PYSA. M. T., K. T., and T. W. conducted the experiments and prepared the manuscript. All authors contributed equally to write and approve the final version of the manuscript.

## Conflicts of interest

There are no conflicts to declare.

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## Notes and references

- 1 F. S. Kim, G. Ren and S. A. Jenekhe, *Chem. Mater.*, 2011, 23, 682–732.



2 L. Zang, Y. Che and J. S. Moore, *Acc. Chem. Res.*, 2008, **41**, 1596–1608.

3 Y. S. Zhao, H. Fu, A. Peng, Y. Ma, Q. Liao and J. Yao, *Acc. Chem. Res.*, 2010, **43**, 409–418.

4 J. Zaumseil and H. Sirringhaus, *Chem. Rev.*, 2007, **107**, 1296–1323.

5 H. Li, B. C. K. Tee, J. J. Cha, Y. Cui, J. W. Chung, S. Y. Lee and Z. Bao, *J. Am. Chem. Soc.*, 2012, **134**, 2760–2765.

6 S. N. Chen, Z. Li, Y. L. Qiao and Y. L. Song, *J. Mater. Chem. C*, 2021, **9**, 1126–1149.

7 T. D. Anthopoulos, S. Setayesh, E. Smits, M. Colle, E. Cantatore, B. de Boer, P. W. M. Blom and D. M. de Leeuw, *Adv. Mater.*, 2006, **18**, 1900–1904.

8 J. Zaumseil, R. H. Friend and H. Sirringhaus, *Nat. Mater.*, 2006, **5**, 69–74.

9 T. D. Anthopoulos, *Appl. Phys. Lett.*, 2007, **91**, 113513.

10 T. Wakahara, P. D'Angelo, K. i. Miyazawa, Y. Nemoto, O. Ito, N. Tanigaki, D. D. C. Bradley and T. D. Anthopoulos, *J. Am. Chem. Soc.*, 2012, **134**, 7204–7206.

11 J. Zhang, H. Geng, T. S. Virk, Y. Zhao, J. Tan, C.-a. Di, W. Xu, K. Singh, W. Hu, Z. Shuai, Y. Liu and D. Zhu, *Adv. Mater.*, 2012, **24**, 2603–2607.

12 Y. Wang, W. Zhu, H. Dong, X. Zhang, R. Li and W. Hu, *Top. Curr. Chem.*, 2016, **374**, 83.

13 J. Zhang, J. Q. Jin, H. X. Xu, Q. C. Zhang and W. Huang, *J. Mater. Chem. C*, 2018, **6**, 3485–3498.

14 Y. Wang, L. J. Sun, C. Wang, F. X. Yang, X. C. Ren, X. T. Zhang, H. L. Dong and W. P. Hu, *Chem. Soc. Rev.*, 2019, **48**, 1492–1530.

15 A. Zampetti, A. Minotto and F. Cacialli, *Adv. Funct. Mater.*, 2019, **29**, 1807623.

16 L. X. Shi, Y. L. Guo, W. P. Hu and Y. Q. Liu, *Mater. Chem. Front.*, 2017, **1**, 2423–2456.

17 T. Wakahara, K. Nagaoka, A. Nakagawa, C. Hirata, Y. Matsushita, K. Miyazawa, O. Ito, Y. Wada, M. Takagi, T. Ishimoto, M. Tachikawa and K. Tsukagoshi, *ACS Appl. Mater. Interfaces*, 2020, **12**, 2878–2883.

18 X. Liu, W. Wang, Z. Fan, W. Huang, L. Luo, C. Yang, J. Zhang, J. Zhao, L. Zhang and W. Huang, *Chem.-Eur. J.*, 2021, **27**, 10448–10455.

19 Z. R. Wang, F. Yu, J. Xie, J. F. Zhao, Y. Zou, Z. P. Wang and Q. H. Zhang, *Chem.-Eur. J.*, 2020, **26**, 3578–3585.

20 A. Mandal, A. Choudhury, P. K. Iyer and P. Mal, *J. Phys. Chem. C*, 2019, **123**, 18198–18206.

21 S. S. Zheng, J. W. Zhong, W. Matsuda, P. Jin, M. Q. Chen, T. Akasaka, K. Tsukagoshi, S. Seki, J. Zhou and X. Lu, *Nano Res.*, 2018, **11**, 1917–1927.

22 M. J. Frisch, G. W. Trucks, H. B. Schlegel, G. E. Scuseria, M. A. Robb, J. R. Cheeseman, G. Scalmani, V. Barone, G. A. Petersson, H. Nakatsuji, X. Li, M. Caricato, A. V. Marenich, J. Bloino, B. G. Janesko, R. Gomperts, B. Mennucci, H. P. Hratchian, J. V. Ortiz, A. F. Izmaylov, J. L. Sonnenberg, D. Williams-Young, F. Ding, F. Lipparini, F. Egidi, J. Goings, B. Peng, A. Petrone, T. Henderson, D. Ranasinghe, V. G. Zakrzewski, J. Gao, N. Rega, G. Zheng, W. Liang, M. Hada, M. Ehara, K. Toyota, R. Fukuda, J. Hasegawa, M. Ishida, T. Nakajima, Y. Honda, O. Kitao, H. Nakai, T. Vreven, K. Throssell, J. A. Montgomery Jr., J. E. Peralta, F. Ogliaro, M. J. Bearpark, J. J. Heyd, E. N. Brothers, K. N. Kudin, V. N. Staroverov, T. A. Keith, R. Kobayashi, J. Normand, K. Raghavachari, A. P. Rendell, J. C. Burant, S. S. Iyengar, J. Tomasi, M. Cossi, J. M. Millam, M. Klene, C. Adamo, R. Cammi, J. W. Ochterski, R. L. Martin, K. Morokuma, O. Farkas, J. B. Foresman, and D. J. Fox, *Gaussian 16, Revision A.03*, Gaussian, Inc., Wallingford CT, 2016.

23 T. Shimazaki, M. M. Hashimoto and T. Maeda, *Proceedings of the 3rd International Workshop on Software Engineering for High Performance Computing in Computational Science and Engineering*, Austin Texas, 2015, vol. 9, DOI: [10.1145/2830168.2830170](https://doi.org/10.1145/2830168.2830170).

24 *Chemcraft – graphical software for visualization of quantum chemistry computations*, <https://www.chemcraftprog.com>, accessed Mar. 24, 2022.

25 T. Xu, N. Chen, Z. He, P. Yu, W. Shen, T. Akasaka and X. Lu, *Chem.-Eur. J.*, 2021, **27**, 10387–10393.

26 M. El Gemayel, M. Treier, C. Musumeci, C. Li, K. Mullen and P. Samori, *J. Am. Chem. Soc.*, 2012, **134**, 2429–2433.

27 Y. Nakajima, M. Hoshino, D. Yamashita and M. Uda, *Advances in Quantum Chemistry*, Academic Press, 2003, vol. 42, pp. 399–405.

28 Y. Nakayama, S. Machida, T. Minari, K. Tsukagishi, Y. Noguchi and H. Ishii, *Appl. Phys. Lett.*, 2008, **93**, 269901.

29 K. Nakano, Y. Kaji and K. Tajima, *ACS Appl. Mater. Interfaces*, 2021, **13**, 28574–28582.

30 D. Mahdaoui, C. Hirata, K. Nagaoka, K. Miyazawa, K. Fujii, T. Ando, M. Abderrabba, O. Ito, M. Takagi, T. Ishimoto, M. Tachikawa, S. Yagyu, Y. B. Liu, Y. Nakajima, Y. Nemoto, K. Tsukagoshi and T. Wakahara, *J. Mater. Chem. C*, 2022, **10**, 3770–3774.

31 I. D. Krysko, A. Y. Freidzon and A. A. Bagaturyants, *Phys. Chem. Chem. Phys.*, 2020, **22**, 3539–3544.

32 T. Shimazaki, Y. Asai and K. Yamashita, *J. Phys. Chem. B*, 2005, **109**, 1295–1303.

