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# Thiophene derivatives as corrosion inhibitors for 2024-T3 aluminum alloy in hydrochloric acid medium

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Thiophene derivatives, namely (*E*)-thiophene-2-carbaldehyde oxime (OXM) and (*E*)-5-(thiophen-2-yl)-1*H*-tetrazole (TET), were synthesized and characterized via <sup>1</sup>H and <sup>13</sup>C NMR. Furthermore, their inhibitory property for AA2024-T3 in 1 M HCl solution was investigated via electrochemical impedance spectroscopy and potentiodynamic polarization at 293 K, together with DFT/B3LYP-based calculations. Numerous global and local descriptors of reactivity such as EHOMO, ELUMO, energy gap, electronegativity ( $\chi$ ), hardness ( $\eta$ ), and frontier molecular orbital repartitions were investigated to describe the reactivity of each molecule. Alternatively, Monte Carlo simulations were performed under the solvation condition on the Al (111) surface to understand the adsorption behavior of the as-studied inhibitors deeply. The inhibition efficiency increased with an increase in the inhibitor concentration, achieving maximum values of 94.0% and 96% at 10<sup>-3</sup> M, respectively. The polarization curves showed that the examined compounds act as mixed-type inhibitors. In addition, the adsorption of these compounds obeyed the Al Awady, Flory-Huggins and Temkin isotherms. The surface characterization analysis via SEM/EDX confirmed the presence of a barrier layer covering the aluminum surface. The experimental inhibition efficiencies were correlated with global descriptors, which confirmed that this theoretical study is useful for the protection of aluminum alloy metal in an acidic medium.

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## 1. Introduction

The use of 2024-T3 aluminum alloy is well known in aerospace, automotive, marine, and chemical applications due to its high strength/weight ratio and low cost.<sup>1–8</sup> Aluminum is generally protected by a natural Al<sub>2</sub>O<sub>3</sub> film on the alloy surface in non-aggressive environments.<sup>4</sup> The oxide layer has many defects or pores and can be easily destroyed by corrosive OH<sup>-</sup> or Cl<sup>-</sup> ions. However, the aluminum alloy AA2024-T3 is very sensitive to localized corrosion in aggressive environments due to the presence of major alloy elements such as copper and magnesium. The microstructure of this alloy is relatively complex and a number of compositionally distinct phases have been identified.<sup>9</sup>

Although possessing favorable mechanical properties, aluminum alloy is relatively susceptible to corrosion and generally requires surface treatment for practical applications.

One of the principal types of second phase particles that are important to the corrosion behavior of an alloy is the S phase (Al<sub>2</sub>CuMg) particle. The surface of the alloy is rich in copper, and thus this phase is responsible for the formation of galvanic microcells between the aluminum matrix, which provides anodic sites, and the copper-rich intermetallic particles, providing cathodic sites.<sup>10–12,15–18</sup> Although aluminum alloy has favorable mechanical properties, it is corroded in the presence of Cl<sup>-</sup>,<sup>19</sup> and this problem remains a concern in different applications.

Currently, various effective protection methods are available, such as anodic oxidation protection,<sup>20</sup> surface coatings, addition of organic inhibitors and plasma electrolytic oxidation protection.<sup>22,23</sup> Among them, inhibitors are the most used and the least expensive in different industries, protecting metals from corrosion via the formation of a protective layer on the surface of the metal. Chromate-based products are widely used for corrosion inhibition in aluminum alloys. However, hexavalent chromates are highly toxic and responsible for many environmental problems.<sup>23</sup> Therefore, chromate-based compounds should be replaced with environmentally friendly inhibitors.

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Organic and inorganic substances can be studied as corrosion inhibitors to protect the AA2024-T3 alloy in chloride electrolytes. Snihirova *et al.* studied the synergistic effects of binary mixtures of Ce-DMTD, 8HQ-SAL, 8HQ-DMTD, and Ce-SAL as environmentally friendly inhibitors for oxygen reduction reactions and demonstrated good inhibition efficiency on AA2024-T3 alloys.<sup>24</sup> They found that the above-mentioned inhibitors adsorbed and precipitated on the surface of the aluminum alloy and blocked the active corrosion sites. Tianbao Zhang *et al.* investigated the corrosion inhibition behavior of an environmentally friendly thiazine-methionine for 2024-T3 aluminum alloy in 1 M acid chloride solution. They observed the formation of a film of compacted adsorptions and an inhibition efficiency of nearly 99%. They also used the electrochemical noise method (ECN), SEM and Kelvin probe force microscopy (KPFM) for a more in-depth study on the localized inhibition performance of AA2024-T3 aluminum alloy.<sup>25</sup> The effect of benzotriazole inhibitors and cerium chloride was studied by Coelho *et al.* to protect and study the behavior of AA2024-T3 aluminum alloy exposed in a neutral solution of NaCl 0.05 M.<sup>26</sup>

The electrochemical results and the surface analysis showed that the formation of a uniform and thin protective film on the surface of the aluminum alloy resulted in high corrosion resistance in AA2024-T3 alloy substrates immersed in a neutral solution in the presence of inhibitors. Another study by Fernine *et al.* used an ecological inhibitor based on *Ocimum basilicum* seed extract (OBSE) for the protection of AA2024 aluminum alloy exposed to a neutral solution of 3 wt% NaCl.<sup>27</sup> Based on EIS measurements, they showed a good corrosion efficiency yield of 95.5% at a concentration of 1 g L<sup>-1</sup>. Polarization curves showed that the OBSE extract provided mixed inhibition, and thus the corrosion of AA2024 was delayed by OBSE.

Inhibitors are commonly used for corrosion protection but are unfavorable toxic compounds. Therefore, various studies have been conducted to develop environmentally friendly, biodegradable, low-toxic, harmless and cheap organic inhibitors that can be used with a sufficient margin of safety. Over the past decade, there have been several joint research reports on the replacement of harmful products with new environmentally friendly inhibitors.<sup>28,29</sup> The importance of organic components as inhibitors is mainly due to the presence of electronegative functional groups as the  $\text{—C=N—}$  group, electronegative nitrogen, sulphur, oxygen atoms and p-electrons in triple or conjugated double bonds in their structure, which are usually good inhibitors.<sup>30</sup>

These elements and sites are fundamental in the action of suitable inhibitors, depending on the specific interaction between the functional groups and the metal surface.<sup>31</sup> Some tetrazole derivatives compounds have been reported, although several studies demonstrate that they are effective corrosion inhibitors for many metals such as copper and its alloys, aluminum and zinc in acid media.<sup>21–24</sup>

Thiophen-based compounds have very promising characteristics, which are related to the presence of the sulfur atom having a larger size compared to the C, N, and O atoms. These atoms play an important role in the inhibitive action of organic compounds against the corrosion of alloys.<sup>32</sup> Various

compounds based on thiophen nuclei are used to reduce copper corrosion in HNO<sub>3</sub> solution. In addition, A. K. Singh *et al.* studied the corrosion inhibition of mild steel using a series of hydrazones of thiophene derivatives in 0.5 M H<sub>2</sub>SO<sub>4</sub> medium.<sup>33</sup>

The aim of this study was to deepen the previous experimental studies by our research team, studying the corrosion inhibition of AA2024-T3 in hydrochloric acid medium using two organic compounds of environmentally friendly thiophen derivatives. Typically, two thiophen derivatives, namely, thiophene-2-carbaldehyde oxime (OXM) and 5-(thiophen-2-yl)-1H-tetrazole (TET), were used as an inhibitors of aluminum alloy in an acidic medium, which is considerably rare in the literature. Furthermore, a theoretical study was conducted using Gaussian software and density functional theory (DFT)<sup>34,35</sup> at the Lee–Yang–Parr correlation functional level (B3LYP) with the 6-311G (d,p) basis sets. The inhibitor adsorption process was also studied and discussed using Monte Carlo simulations. DFT calculations and Monte Carlo simulations allowed us to study the effect of the structural property of the inhibitor in detail according to its inhibitory efficiency.

In this study, the adsorption energy ( $E_{\text{ads}}$ ) of the neutral and protonated forms of each thiophen derivative molecule was investigated on the Al (111) surface in aqueous phases.

## 2. Material and methods

### 2.1 Preparation of the materials and solutions

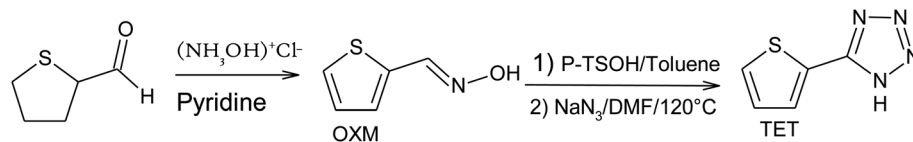
AA2024-T3 was utilized as the substrate in the experiments. The chemical composition of the substrate in weight% is as follows: Cu, 4.18; Mg, 1.3–1.8; Mn, 0.3; Si, 0.5; Fe, 0.5; Zn, 0.3; and Al, 93.52. Each substrate was abraded with 120-, 240-, 400-, 600-, 800-, 1200-order emery papers, washed with distilled water and acetone and dried at room temperature. All measurements were performed in 1 M HCl acid solution at 25 °C. The studied inhibitors prepared by immersing them in aggressive solution in the concentration range of 10<sup>-3</sup> to 10<sup>-6</sup> M, and we used 3 mL of methanol to solubilize the products before adding the acid.

### 2.2 Synthesis of OXM and TET

The two organic compounds synthesized were (5-(thiophen-2-yl)-1H-tetrazole) (TET) substituted in position 5 by an electron withdrawing group, and thiophene-2-carbaldehyde oxime (OXM).<sup>14</sup> The following reaction scheme summarizes the steps of this synthesis (Scheme 1).

**2.2.1 (E)-Thiophene-2-carbaldehyde oxime (OXM).** Oximes are formed *via* the well-known addition elimination mechanism, which is usually catalyzed by acids. The general method involves stirring the aldehyde with excess hydroxylamine hydrochloride in the presence of sodium hydroxide in an aqueous medium. In our case, hydroxylamine hydrochloride was treated with 1.5 equivalent of NaOH (6 N) and 0.9 equivalent of aldehyde, which after 48 h of stirring at room temperature, led to the corresponding oxime with a yield of around 70%. Thus, the use of pyridine as the base and reaction solvent led to a better result. Stirring for 2 h in refluxing pyridine led to the formation of this oxime in excellent yield. The compound yield





Scheme 1 Synthetic route for the tested compounds.

= 88% (white solid); m.p. = 132–134 °C.  $^1\text{H}$  NMR (300.13 MHz,  $\text{CDCl}_3$ )  $\delta_{\text{H}}$ , ppm: 6.53 (t, 1H, =CH-,  $3J = 3.7$  Hz), 6.73 (d, 1H, =CH-,  $3J = 3.7$  Hz), 6.85 (d, 1H, =CH-,  $3J = 3.7$  Hz), 7.05 (s, 1H, -CH=N), 8.58 (e, 1H, =N-OH).  $^{13}\text{C}$ -NMR (75.47 MHz,  $\text{CDCl}_3$ )  $\delta_{\text{C}}$ , ppm: 123.2 (1C, =CH-), 124.5 (1C, -CH-), 126.1 (1C, -CH-), 140.6 (1C, =C<sub>thioph</sub>), 150.1 (1C, -CH=N).

**2.2.2 (E)-5-(Thiophen-2-yl)-1H-tetrazole (TET).** The addition of one equivalent of *p*-toluenesulfonic acid (TsOH) to the oxime (OXM) in toluene resulted in the corresponding tosylate being obtained in a yield of 85%. The latter was heated with one equivalent of sodium azide in DMF for 12 h at 120 °C. Our target product (TET) was purified *via* recrystallization from ethyl acetate with a good yield. The compound yield = 70% (white solid); m.p. = 98–100 °C.  $^1\text{H}$ -NMR (300.13 MHz,  $\text{CDCl}_3$ )  $\delta_{\text{H}}$ , ppm: 6.17 (e, 1H, -NH<sub>tetraz</sub>), 6.61 (t, 1H, =CH-,  $3J = 3.6$  Hz), 6.79 (d, 1H, =CH-,  $3J = 3.6$  Hz), 6.82 (d, 1H, =CH-,  $3J = 3.7$  Hz).  $^{13}\text{C}$ -NMR (75.47 MHz,  $\text{CDCl}_3$ )  $\delta_{\text{C}}$ , ppm: 124.2 (1C, =CH-), 124.7 (1C, =CH-), 126.2 (1C, =CH-), 141.7 (1C, =C<sub>thioph</sub>), 161.3 (1C, =C<sub>tetraz</sub>).

### 2.3 Electrochemical methods

Electrochemical tests were carried out using a potentiostat (VersaSTAT 4) and analyzed with the Versa studio software. The electrochemical set-up utilized in all manipulations contained three electrodes, *i.e.*, the AA2024 substrate (working electrode), a counter electrode (platinum electrode) and an Ag/AgCl electrode (reference electrode), and their surface was immersed in electrolyte with an area of 1 cm<sup>2</sup>.

The surface was immersed in 1 M HCl without and with the inhibitors. Potentiodynamic polarization measurements were executed automatically from -800 to -100 mV *vs.* OCP at a scan rate of 1 mV s<sup>-1</sup>. The stabilization time of the surface in solution was 30 min.

The impedance results were analyzed using Nyquist and Bode plots to study the corrosion characteristics at room temperature. The EIS tests were executed by varying the frequency range (100 kHz to 10 MHz) with a low amplitude disturbance of 10 mV. The EIS parameters were adjusted using an electrical circuit extracted by the Z-View software.

### 2.4 Surface characterization

The surface morphologies of AA2024-T3 were analyzed *via* SEM-coupled energy dispersive X-ray spectroscopy (EDX) using an FEI Quanta 200 SEM instrument equipped with an EDX analyzer. The substrates were immersed for 24 h at 25 °C in 1 M HCl solution without and with inhibitor.

### 2.5 Quantum chemical calculations

To predict the inhibition reactivity to the aluminum alloy surface, theoretical calculations were performed using density functional theory (DFT). The geometric optimizations were performed using the three Becke functional parameters (B3), combined with functional correlation (B-3LYP),<sup>13</sup> and combined with the 6-311G (d, p) basis set using the Gaussian 09 software.<sup>26</sup> DFT was used to calculate the global and local molecular reactivity parameters, such as highest occupied molecular orbital energy level ( $E_{\text{HOMO}}$ ) and lowest unoccupied molecular orbital energy level ( $E_{\text{LUMO}}$ ), and the other quantum parameters are presented by the following mathematical formulas:

$$\text{IP} = -E_{\text{HOMO}} \quad (1)$$

$$\text{EA} = -E_{\text{LUMO}} \quad (2)$$

$$\chi = \frac{\text{IP} + \text{EA}}{2} \quad (3)$$

$$\eta = \frac{\text{IP} - \text{EA}}{2} \quad (4)$$

$$\sigma = \frac{1}{\eta} \quad (5)$$

The expression for a transferred charge fraction ( $\Delta N$ ) can be used to evaluate the tendency of a molecule to donate or accept electrons from a metal with eqn (6), as follows:

$$\Delta N = \frac{\varphi_{\text{Al}} - \chi_{\text{inh}}}{2(\eta_{\text{Al}} - \eta_{\text{inh}})} \quad (6)$$

where  $\varphi_{\text{Al}}$  equals 4.82 eV and  $\eta_{\text{Al}}$  equals 0.0 eV,<sup>36</sup> which are the work function and the absolute hardness of the aluminum metal, respectively.

The expression for the electrophilicity index ( $\omega$ ) can be utilized to determine the ability of a molecule to generate electron transfer, as follows:

$$\omega = \frac{\chi^2}{2\eta} \quad (7)$$

### 2.6 Monte Carlo simulation details

Monte Carlo simulation was performed using the Adsorption Locator module in the Materials Studio 8.0 software. This method was investigated to compute the low configuration adsorption energy of the interactions of both xanthene derivatives with an iron surface.<sup>20</sup> The COMPASS force field was used to optimize the structure of all the components of the system at 25 °C.<sup>21</sup> Moreover,



the simulation study for the gas phase was carried out on an Al (111) crystal with a slab of 5 Å, enlarged (8 × 8) supercell and the vacuum slab was built above the aluminum plane with a thickness of 30 Å.<sup>22</sup> In the aqueous phase, 150 molecules of water were added to the simulation system, enlarging the supercell to (12 × 12) with a vacuum slab of 50 Å thickness, where the supercell enlarged to accommodate the water molecules.<sup>23</sup>

### 3. Results and discussion

#### 3.1 Open circuit potential and potentiodynamic polarization study

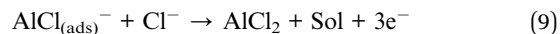
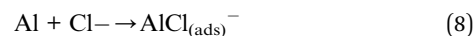
The open-circuit potential (OCP) sweep primarily represents the process of changing the electrode from unstable to stable for 1800 s. The OCP as a function of time was recorded to understand the characteristics of the deposition process of TET and OXM on the AA2024-T3 surface. The OCP curves as a function of time are shown in Fig. 1, where the waiting period of 1800 s was sufficient for the test.

#### 3.2 Electrochemical polarization measurements

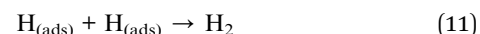
The electrochemical polarization test was executed to thoroughly investigate the mechanism of OXM and TET protecting the surface of AA2024-T3. Fig. 2 shows the polarization curves of the AA2024-T3 electrode immersed in 1 M HCl containing

different concentrations of OXM and TET at 298 K. According to the literature, the cathodic and anodic branches of AA2024-T3 in HCl can be explained by the proposed reaction mechanism, as follows:<sup>37</sup>

Anodic reactions:



Cathodic reactions:



In eqn (8) and (9), it is obvious that aluminum readily reacts to form  $\text{AlCl}_{(\text{ads})}^-$  and  $\text{AlCl}_2^+$  in 1 M HCl solution.

The electrochemical parameters were calculated by extrapolating the Tafel region from the cathodic curve. The results in Fig. 2 show a decrease in all the anodic and cathodic current densities, where the corrosion potentials are below 85 mV. This suggests that the TET and OXM inhibitors control HCl aggression *via* a mixed-type mechanism (the  $E_{\text{corr}}$  differences are below ± 85 mV). It was observed that the cathodic branches are almost parallel for both inhibitors, which suggests that the TET

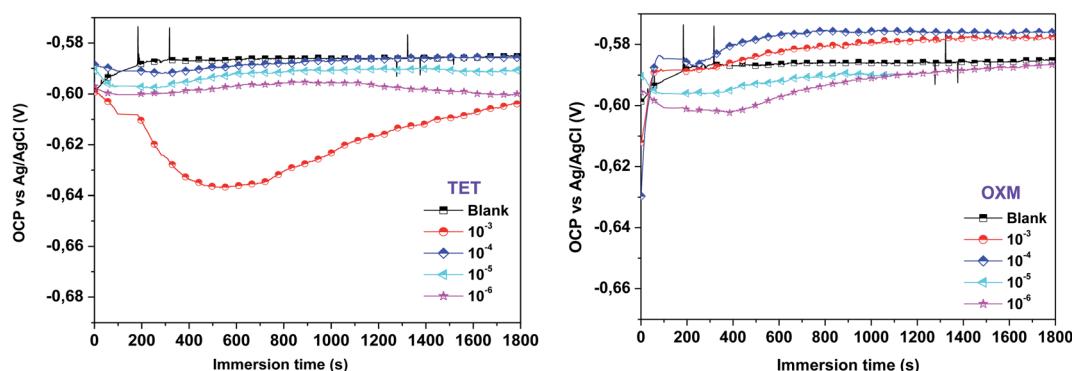


Fig. 1 OCP-time curve of the AA2024-T3 electrode immersed in 1 M HCl solution containing different concentrations of TET and OXM.

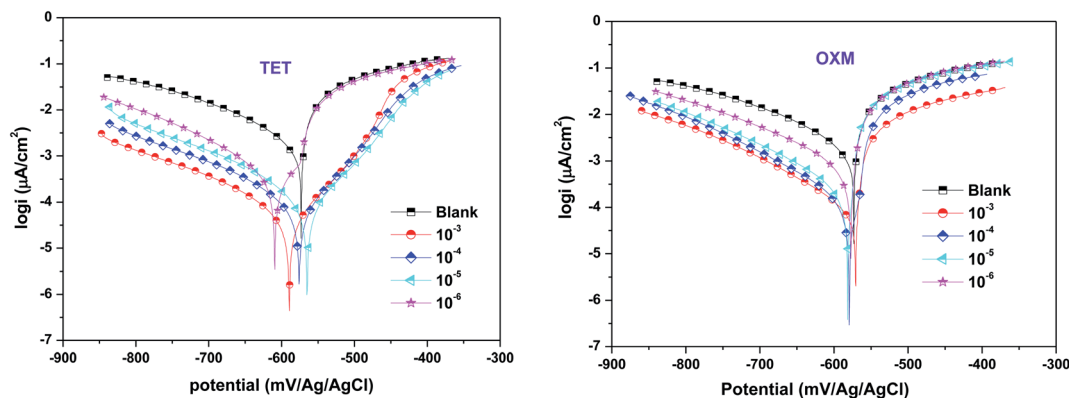


Fig. 2 Polarization plots of AA2024-T3 electrode in 1 M HCl and protected with different concentrations of TET and OXM at 298 K.



and OXM inhibitors do not change the process of the cathodic branches but just decrease the rate of reaction to delay corrosion.<sup>38</sup> Thus, both inhibitors affected the shape of the anodic regions. Moreover, the addition of TET led to the appearance of pseudo-bearing, which demonstrates that the inhibitors have an effect on the anodic zone. The parameters of the PDP curves are presented in Table 1, including  $E_{\text{corr}}$ ,  $i_{\text{corr}}$ ,  $\beta_c$ ,  $\beta_a$  and  $\eta_{\text{Tafel}}$ . As shown in Table 1, the values of the potential differences of TET and OXM added to  $E_{\text{corr}}$  are 30 mV and 19 mV at 298 K, respectively. All the values are below 85 mV, indicating that TET and OXM can be classified as mixed inhibitors.<sup>39</sup>

In addition, the current density values decreased with an increase in the concentration of the TET and OXM inhibitors, which reached about  $90 \mu\text{A cm}^{-2}$  and  $116 \mu\text{A cm}^{-2}$  for the concentration of  $10^{-3}$  M, respectively. Consequently, the  $\eta$  efficiencies increased for both inhibitors to reach maximum corrosion inhibition of 96% for TET and 94% for OXM at  $10^{-3}$  M. This indicates that TET and OXM can be considered effective inhibitors and can maintain strong adsorption over a wide concentration range.

### 3.3 Electrochemical impedance measurements

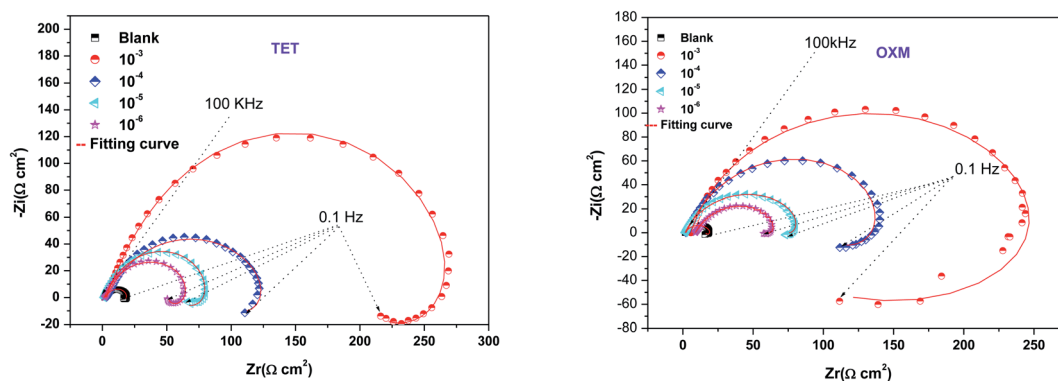
Impedance experiments were conducted to better understand the electrochemical processes occurring at the interface of the material/solutions, and also the dynamic process of the protective layer phenomenon. The influence of different

concentrations of TET and OXM on their inhibitory efficiency was examined by means of EIS.

Fig. 3 shows the Nyquist diagrams of the AA2024-T3 electrode in 1 M HCl without inhibitor and with inhibitor (Fig. 4). According to Fig. 2, two time constants can be observed, namely, a depressed capacitive time constant at high-frequency values and an inductive time constant at low frequencies. Similar behavior was observed by A. Yurt *et al.*<sup>40</sup> In these spectra, at high frequencies, the depressed capacitive loop is attributed to the corrosion charge transfer process, which depending on the depressive character of the semicircle, is mostly assigned to an inhomogeneous metal surface. The second time constant at low frequencies (inductive loop) is often related to the surface relaxation of species or the adsorption of the intermediate products of the corrosion reaction on the surface (TET and OXM), as well as the reactive products. The addition of TET and OXM did not change the shape of the capacitive loop at high frequency. In addition, there were inductive loops in the low-frequency region, which are well defined at a concentration of  $10^{-3}$  M for both synthesized inhibitors. As the inhibitor concentration increased, the size of the loops also increased. This results in the restructuring of the passivity film, leading to better resistance to charge transfer occurring in the outer layer of the resulting film. Table 2 presents the EIS data for the 2024-T3 aluminum alloy electrode in 1 M HCl and protected with different concentrations of TET and OXM at 298 K. The EIS parameters of these loops were fitted with an electrical

**Table 1** The polarization data of the AA2024-T3 electrode in 1 M HCl and protected with different concentrations of TET and OXM at 298 K

Medium	Conc. [ $\text{mol L}^{-1}$ ]	$E_{\text{corr}}$ [mV/Ag/AgCl]	$i_{\text{corr}}$ [ $\mu\text{A cm}^{-2}$ ]	$-\beta_a$ [mV dec $^{-1}$ ]	$-\beta_c$ [mV dec $^{-1}$ ]	$\eta_{\text{Tafel}}\%$
1 M HCl		−574	2888	35	183	
TET	$10^{-6}$	−606	558	37	117	80
	$10^{-5}$	−591	316	21	127	89
	$10^{-4}$	−584	206	18	104	92
	$10^{-3}$	−607	90	26	119	96
OXM	$10^{-6}$	−736	651	16	139	77
	$10^{-5}$	−581	334	14	161	88
	$10^{-4}$	−583	308	21	136	89
	$10^{-3}$	−594	116	15	128	94



**Fig. 3** Nyquist diagrams of AA2024-T3 electrode in 1 M HCl with various concentrations of TET and OXM at 298 K.





equivalent circuit, as displayed in Fig. 3. This circuit allows the detection of two time constants in the absence and presence of inhibitors. The first time constant is due to the capacitive loop and the second time constant is due to the inductive loop. The equivalent circuit contains the solution resistance ( $R_s$ ), charge transfer resistance ( $R_{ct}$ ), inductive element resistance (RL), inductive element coil ( $L$ ) and CPE, which is the constant phase angle element ( $Q$ ) and can be obtained using the following expression:

$$Z_{CPE} = \frac{1}{Y_0(j\omega)^n} \quad (12)$$

where  $Y_0$  is the constant phase angle element of CPE,  $n$  is the dispersion effect index exponent, which provides a measure of the unevenness of the electrode surface and its range is between ( $-1 \leq n \leq 1$ ),  $j$  is the unit of the imaginary frequency, and  $\omega$  is the angular frequency in  $\text{rad s}^{-1}$  equal to  $2\pi f$ . The formulas for the calculation of the double layer capacitance ( $C_{dl}$ ) and the inhibition efficiency ( $\eta$ ) are as follows:<sup>41</sup>

$$C = Y_0(\omega)^{n-1} = Y_0(2\pi f_{Zim-Max}) \quad (13)$$

$$\eta(\%) = \frac{R_{ct} - R_{ct,0}}{R_{ct}} \times 100 \quad (14)$$

where  $f_{Zim-Max}$  is the frequency corresponding to the maximum imaginary part of the impedance,  $R_{ct}$  is the charge transfer resistance with inhibitor, and  $R_{ct,0}$  is the charge transfer resistance without inhibitor.

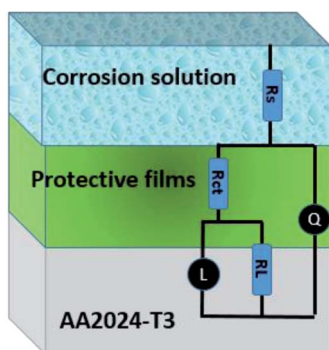


Fig. 4 Equivalent circuit used to simulate the EIS diagram.

According to Table 2, it can be concluded that with an increase in the concentration of TET and OXM, there was an increase in the charge transfer resistance markedly to a value of 226.8 ( $\Omega \text{ cm}^{-2}$ ) for TET and 214.9 ( $\Omega \text{ cm}^{-2}$ ) for OXM up to  $10^{-3}$  M. In contrast, the values of the double layer ( $C_{dl}$ ) dropped rapidly with an increase in the concentration of the TET and OXM inhibitors, suggesting that the water molecules adsorbed on the surface of AA2024 were replaced by molecules of the synthesized inhibitors, TET and OXM, with a lower dielectric constant according to the Helmholtz model equation, as follows:<sup>42</sup>

$$C_{dl} = \frac{\epsilon^0 \epsilon}{d} S \quad (15)$$

where  $\epsilon$  is the local dielectric constant,  $\epsilon^0$  is the dielectric constant of air,  $S$  is the surface of the electrode, and  $d$  is the thickness of the electrical double layer.

Fig. 5 shows the Bode plots of the AA2024 electrode with various TET and OXM corrosion concentrations at 298 K, showing the exact frequencies at which the data points were recorded, which explains the specific impedance behavior at the system frequency. Fig. 3 shows the Bode magnitude variation and Bode phase angle diagrams for the corrosion of the 2024 aluminum alloy in 1 M HCl. As the concentration of the TET and OXM inhibitors increased, the phase angle plots became wider in the high frequency range, which indicates that the corrosion inhibition of AA2024 was enhanced due to the increase in the inhibitor coverage on the aluminum alloy. Also, the impedance modulus was the highest at the concentration of  $10^{-3}$  M, indicating the very high efficiency of the adsorption of the inhibitors at the AA2024 surface. In fact, this demonstrates that there was better protection against corrosion in the corrosive HCl medium.

### 3.4 Surface analysis (SEM/EDX)

Fig. 5 shows the scanning electron microscopy (SEM) images of the 2024-T3 aluminum alloy in 1 M HCl and in the presence of the TET and OXM inhibitors with the optimal concentration of  $10^{-3}$  M. It can be seen in Fig. 6(A) that the sample surface with no inhibitors was affected and damaged due to the corrosive ions of  $\text{H}^+$  and  $\text{Cl}^-$  on the surface of the metal, which shows that the sample was intensely corroded when it was in contact with a corrosive acid. In addition, the surface was covered with remarkable corrosion products due to the dissolution of the metal and the creation of corrosion pits.<sup>43</sup>

Table 2 Impedance parameters for AA2024-T3 in 1 M HCl in various concentrations TET and OXM at 298 K

Medium	Conc. [mol/]	$R_s$ ( $\Omega \text{ cm}^{-2}$ )	$R_{ct}$ ( $\Omega \text{ cm}^{-2}$ )	CPE ( $\mu\text{F}\cdot\text{s}^{n-1}$ )	$n_{dl}$	$C_{dl}$ ( $\mu\text{F}\cdot\text{cm}^{-2}$ )	RL	$R_p\%$	$L$	$\eta\%$
1 M HCl		1.481	15.3	1979	0.847	1054.85	5.8	21.1	0.723	—
TET	$10^{-6}$	1.769	53.8	804	0.762	302.48	12.4	65.7	2.554	67
	$10^{-5}$	1.301	69.3	481	0.765	170.29	30.6	99.9	3.217	78
	$10^{-4}$	2.236	103.2	304	0.784	117.93	57.4	160.6	16.99	86
	$10^{-3}$	2.555	226.8	211	0.779	89.78	121.8	388.6	17.99	94
OXM	$10^{-6}$	9.047	47.7	823	0.774	320.83	13.1	60.9	5.101	65
	$10^{-5}$	1.571	74.7	519	0.773	200.31	18.0	92.8	2.541	77
	$10^{-4}$	3.214	67.9	367	0.838	179.86	63.0	130.9	32.64	83
	$10^{-3}$	5.668	214.9	194	0.836	104.46	51.4	266.3	15.1	91



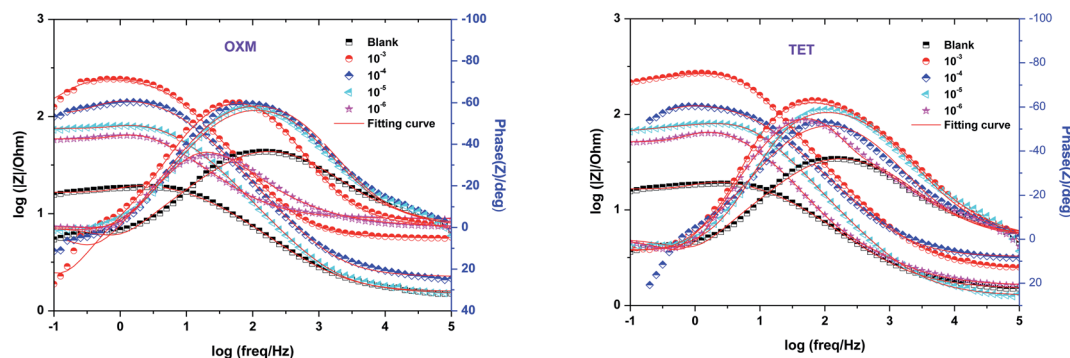


Fig. 5 Bode diagrams of AA2024-T3 electrode in 1 M HCl with various concentrations of TET and OXM at 298 K.

However, the surface morphology of the 2024-T3 aluminum alloy improved in the presence of the TET and OXM inhibitors at their optimum concentration, confirming that the TET and OXM molecules formed a protective layer on the surface of AA2024, which protected the metal from the corrosive environment.<sup>44</sup>

The EDX data allowed the characterization of the elemental composition of the surface of the aluminum alloy after immersion in the corrosive and inhibited solutions for a certain time. The EDX spectrum and Table 3 show the elemental composition of the aluminum matrix in the HCl medium without the inhibitors, which contained 40.17% Al, 13.18% Cu and 34.80% O. The presence of oxygen is due to the formation of an oxide film, which was formed by the slow atmospheric oxidation of aluminum during the SEM/EDX analysis. Also, 6.93% chloride atoms was present in the uninhibited solution.

These results can be explained by the fact that the corrosive hydrochloric acid solution reacted with the aluminum, which suggests the strong dissolution of AA2024 (Fig. 6). In the presence of TET and OXM, the percentage of aluminum was higher, *i.e.*, 60.87% and 62.67%, respectively, and the Cl and O signals decreased in the presence of OXM and disappeared in the presence of TET.<sup>45</sup> This is because the TET inhibitor reduced the attack of chloride ions and oxidation of aluminum on the metal surface by adsorption and the formation of a protective layer on the metal surface. In addition, the percentage of carbon was higher in the presence of the inhibitors.<sup>46</sup>

### 3.5 Isotherm and adsorption parameter

According to these results, it can be suggested that the studied inhibitors follow the El Awady, Flory-Huggins and Temkin

isotherms given that their  $R^2$  coefficients are close to unity.<sup>16,47,48</sup> Furthermore, the values of  $K_{ads}$  shown in Table 4 have no significance in the case of the Frumkin and Freundlich isotherms, which indicates that the studied molecules do not obey these models although their regression coefficient  $R^2$  values are close to unity.<sup>49</sup> However, the A parameter has a negative value, which implies the presence of a repulsive interaction.<sup>50</sup>

According to the literature, the electrostatic interaction (physical adsorption) occurs at the charged metal/solution interface when  $\Delta G_{ads}$  is around  $-20 \text{ kJ mol}^{-1}$ . However, when this value is greater than  $-40 \text{ kJ mol}^{-1}$ , chemisorption occurs, which means that a coordination bond has been created between the inhibitor and the aluminum surface.<sup>51,52</sup> In the present investigation, the obtained value of  $\Delta G_{ads}$  using the three isotherm models reflects the chemical adsorption behavior of the studied products on the working electrode surface, forming strong bonds (Fig. 7).

## 4. Quantum chemistry results

### 4.1 Structural reactivity of OXM and TET

Theoretical methods are one of the most efficient ways to understand the inhibition process based on the determination of several chemical parameters.<sup>64,65</sup>

To find the real form existing in solution, we used the MarvinSketch program before starting the theoretical study. According to the result obtained (Fig. 8), it can be concluded that the TET inhibitor exists in neutral form, whereas OXM is in the protonated form. Consequently, we ran DFT and Monte Carlo simulations in these obtained forms at pH = 0.

Table 3 Percentage atomic and mass contents of the elements obtained from EDX spectra

Adsorbed elements		Al	Cu	Cl	O	C
AA2024 in 1 M of HCl	Mass%	40.17 ± 0.43	13.18 ± 0.34	6.93 ± 0.23	34.80 ± 0.35	4.90 ± 0.12
	Atom%	33.27 ± 0.36	4.64 ± 0.12	4.37 ± 0.14	48.60 ± 0.50	9.12 ± 0.23
AA2024 in 10 <sup>-3</sup> M of OXM	Mass%	60.87 ± 0.42	3.28 ± 0.24	5.15 ± 0.14	18.32 ± 0.41	12.39 ± 0.41
	Atom%	48.74 ± 0.34	1.11 ± 0.08	3.14 ± 0.09	24.73 ± 0.56	22.28 ± 0.74
AA2024 in 10 <sup>-3</sup> M of TET	Mass%	62.67 ± 0.41	4.10 ± 0.25	—	20.13 ± 0.40	13.10 ± 0.37
	Atom%	49.05 ± 0.32	1.36 ± 0.08	—	26.56 ± 0.52	23.03 ± 0.66



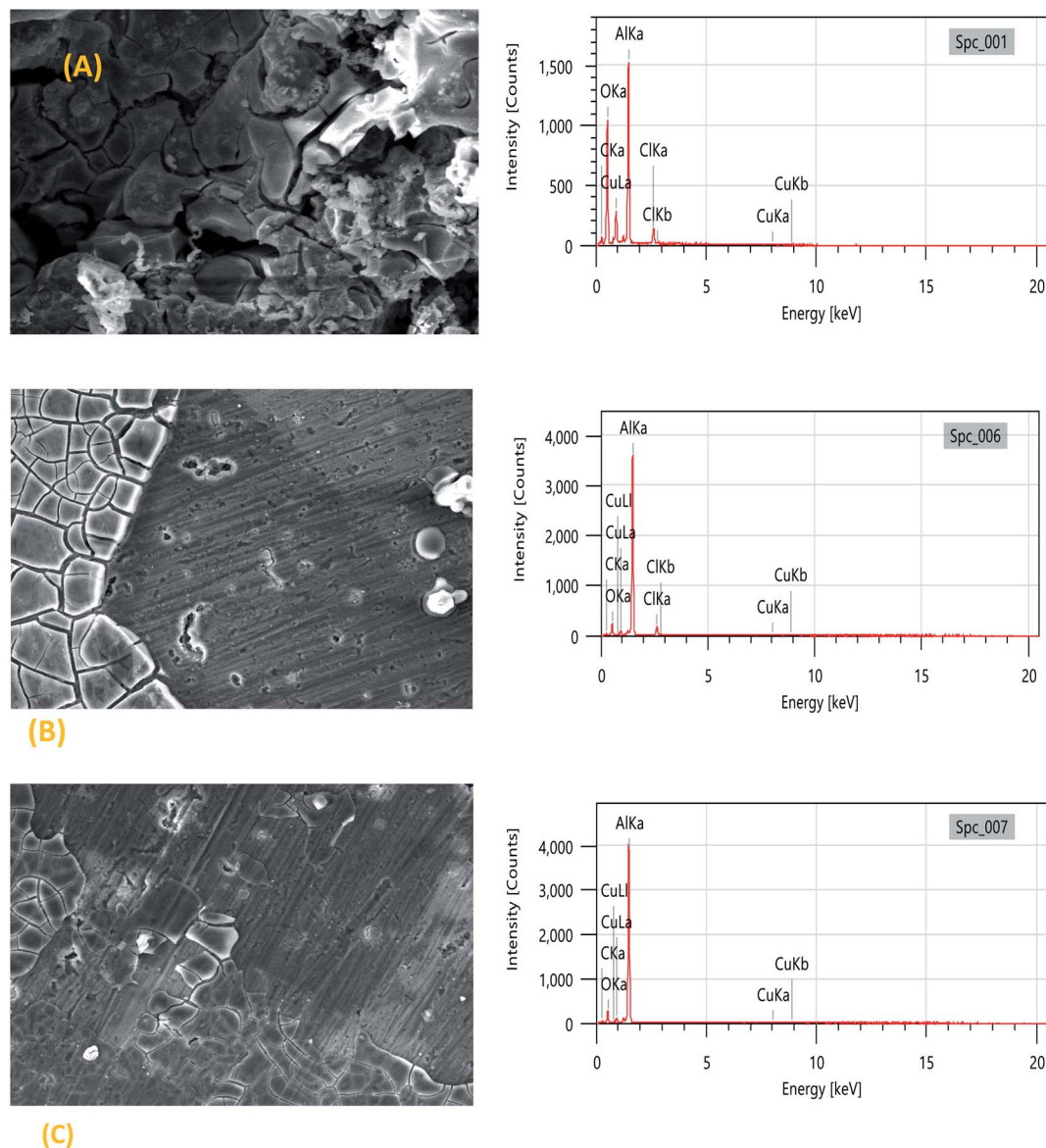


Fig. 6 SEM micrographs and EDX plots for (A) blank, (B) OXM and (C) TET.

Table 4 Parameters obtained from the various isotherm models tested in this study

Isotherm	Inhibitor	$R^2$	Parameters		$K$	$\Delta G_{\text{ads}}^\circ$ (kJ mol <sup>-1</sup> )
Freundlich	TET	0.99	$Z$	17.96	1.39E+00	$-1.0770 \times 10^4$
	OXM	0.98		18.12	1.34E+00	$-1.0679 \times 10^4$
El-Awady	TET	0.99	1/y	3.42	$4.84 \times 10^6$	$-4.81 \times 10^4$
	OXM	0.99		4.04	$5.37 \times 10^6$	$-4.84 \times 10^4$
Frumkin	TET	0.98	$D$	-3.35	$2.20 \times 10^{-2}$	$-4.99 \times 10^2$
	OXM	0.99		-2.97	$3.83 \times 10^{-2}$	$-1.86 \times 10^3$
Flory-Huggins	TET	0.98	$X$	3.86	$1.88 \times 10^7$	$-5.15 \times 10^4$
	OXM	0.99		4.79	$4.11 \times 10^7$	$-5.34 \times 10^4$
Temkin	TET	0.99	$A$	-11.60	$2.84 \times 10^{12}$	$-8.11 \times 10^4$
	OXM	0.99		-12.20	$3.36 \times 10^{12}$	$-8.15 \times 10^4$

The computational approach was performed using DFT and the B3LYP 6-311G(d,p) basis set. Some of the descriptors were directly obtained from Gaussian output files including  $E_{\text{HOMO}}$ ,

$E_{\text{LUMO}}$ , and  $\mu$ . However, others were calculated separately ( $\Delta E$ ,  $\Delta N$ ,  $\eta$ ,  $\chi$  and  $\sigma$ ). The frontier molecular orbitals (HOMO and LUMO), the electrostatic potential maps (ESPMs) and the





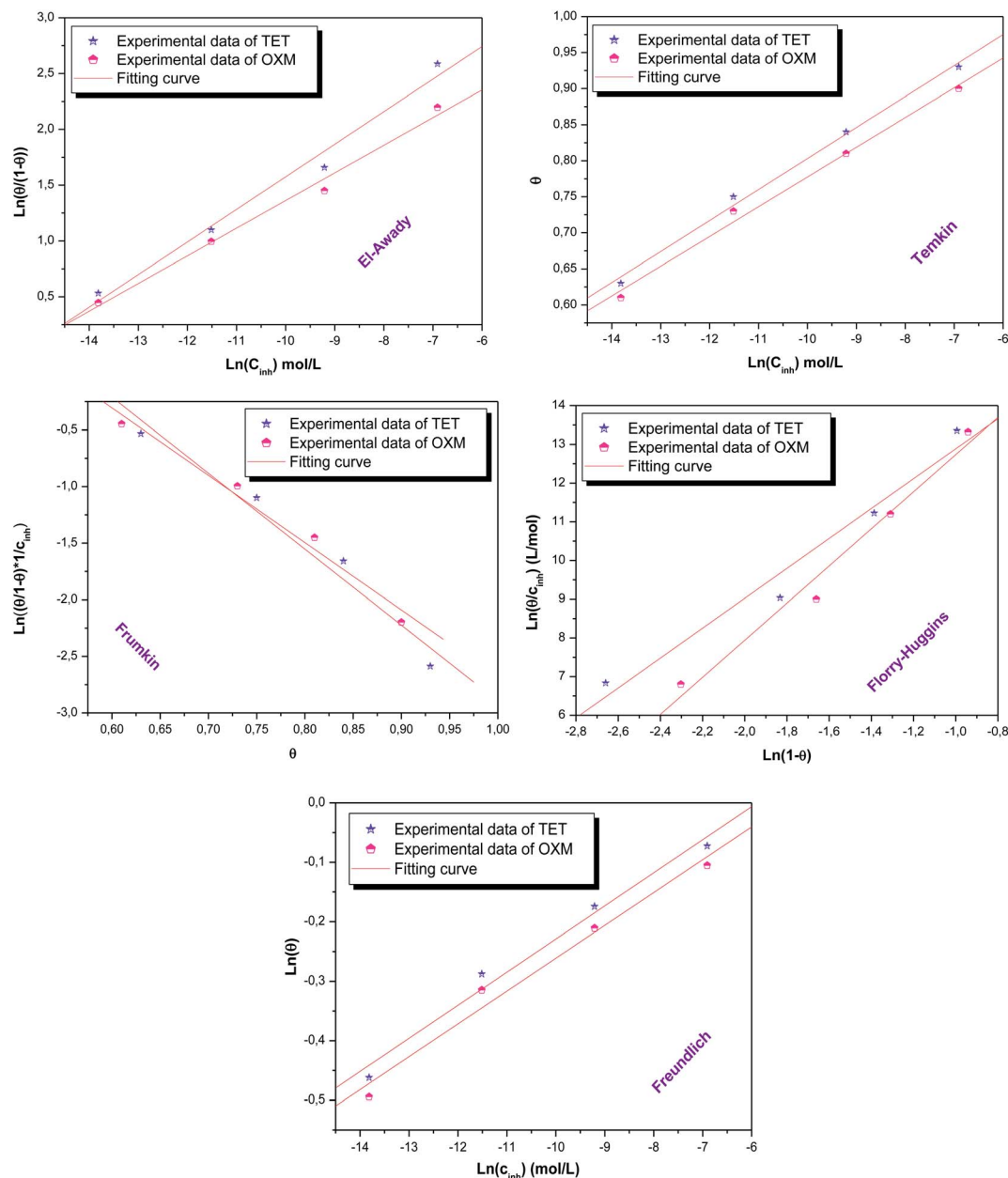


Fig. 7 Isotherms models tested for OXM and TET at 298 K.

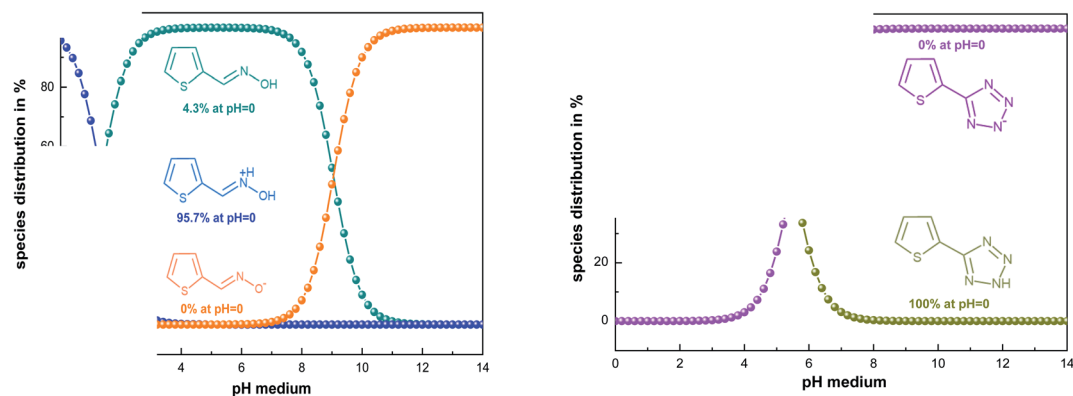


Fig. 8 Physicochemical analysis of OXM and TET inhibitors using the MarvinSketch software.

Table 5 Quantum chemical descriptors of OXM and TET in aqueous solution

Molecule	$E_{\text{HOMO}}$ (eV)	$E_{\text{LUMO}}$ (eV)	$\Delta E_{\text{g}}$ (eV)	$\sigma$ (eV <sup>1</sup> )	$\chi$ (eV)	$\Delta N$ (eV)	$\mu$ (D)	$\omega$	$\varepsilon$
Protonated OXM	−7.959	−3.019	4.940	0.404	5.489	−0.135	4.657	6.099	0.163
TET	−6.753	−1.845	4.907	0.407	4.299	0.105	7.677	3.766	0.265

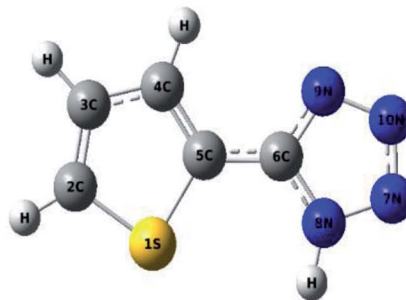
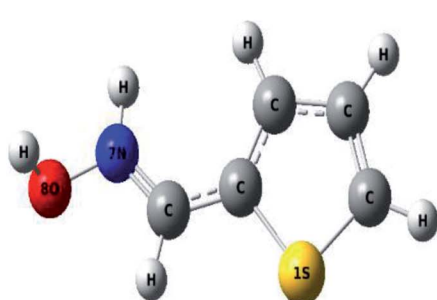
optimized structure of the studied inhibitor are represented in Fig. 8. The global quantum descriptors are listed in Table 5.

It can be seen from Fig. 9 that the density of the HOMO and LUMO is mainly distributed at the edges of the molecules (Fig. 10). In addition, it can be noticed from the MEP that the red regions in the molecular electrostatic potential (MEP) refer to the negative electrostatic potential and are intensified around the oxygen atoms of OXM and nitrogen atoms of TET.<sup>53</sup> This

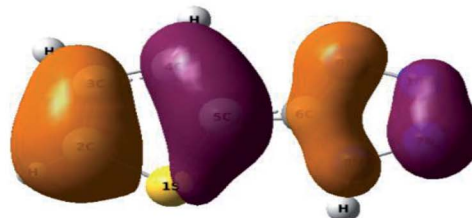
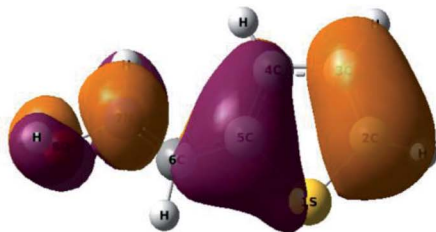
indicates that the studied compounds were adsorbed on the surface of the studied material.<sup>54</sup>

According to the literature, global descriptors can clarify the ability to accept and release electrons.<sup>55,56</sup> As shown in Table 5, the small gap energy ( $\Delta E_{\text{gap}}$ ) value for the TET inhibitor can offer higher reactivity.<sup>57</sup> Moreover, the dipole moment ( $\mu$ ) is another variable that can also provide information about the reactivity of a molecule. Generally, this parameter increases with an increase in the inhibition performance obtained

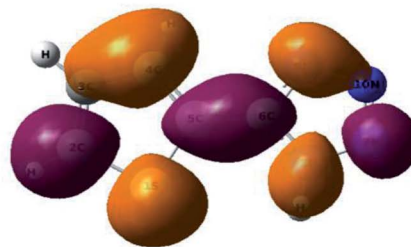
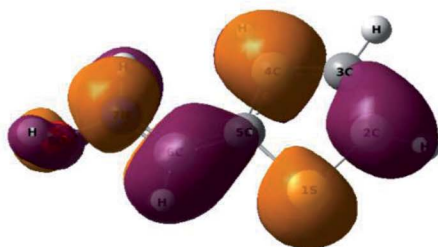
### Optimized structure



### HOMO



### LUMO



### Protonated OXM

### Neutral TET

Fig. 9 Optimized structures of the HOMO and LUMO protonated OXM and neutral TET molecules by the DFT method at the B3LYP/6-311G (d,p) level.



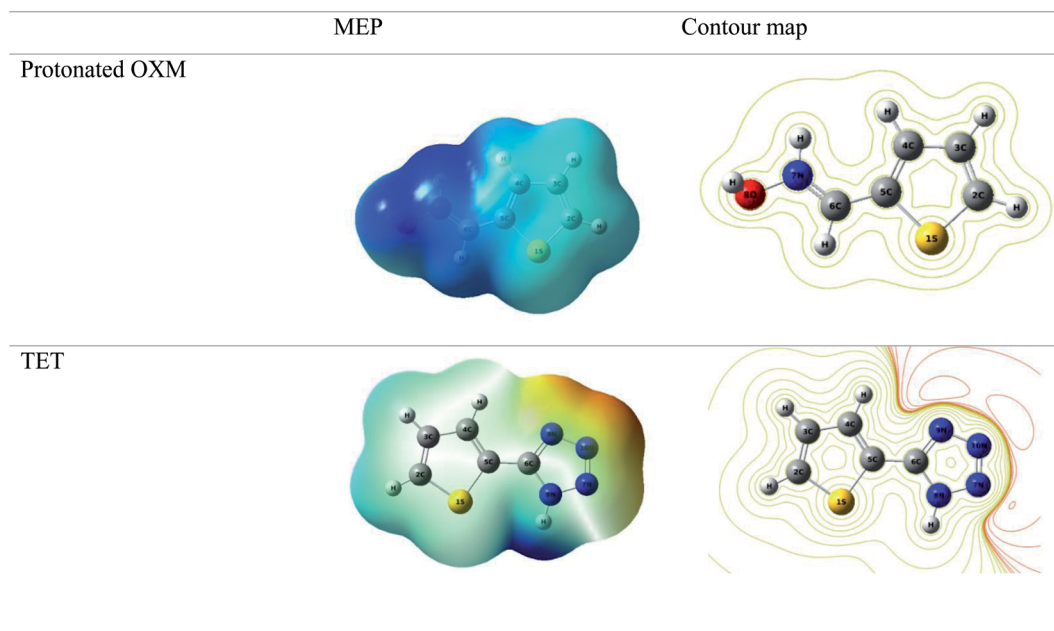


Fig. 10 Spatial distribution of OXM and TET.

experimentally. However, the opposite behavior was found in others works. In the present investigation, the weak value of the dipole moment (4.657D) indicates that the adsorption of the

TET inhibitor on the surface of aluminum resulted in better inhibition efficiency.<sup>58</sup> Furthermore, the fraction of electrons transferred ( $\Delta N$ ) quantifies the transfer of electrons from the

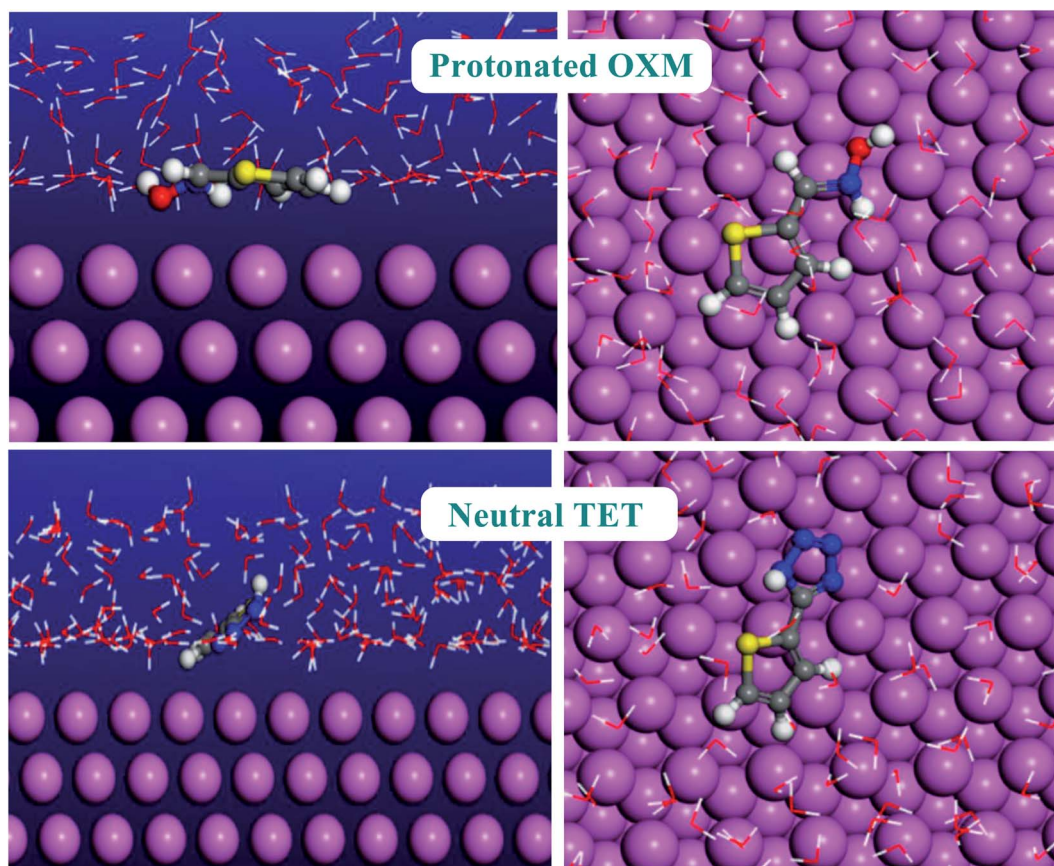


Fig. 11 Most stable configuration systems obtained by Monte Carlo simulation.

**Table 6** Adsorption energy parameters of the stable configuration system (all units in kcal mol<sup>-1</sup>)

Systems	Adsorption energy of inhibitor	Adsorption energy of water molecules
Al (111)/OXM/150H <sub>2</sub> O	−2359	−11
Al (111)/TET/150H <sub>2</sub> O	−2376	−10

molecule to metal if  $\Delta N > 0$  and from the metal to molecule if  $\Delta N < 0$ .<sup>59,60</sup> Thus, it can be concluded that the quantum global descriptors signify that the thiophene inhibitors exhibit high reactivity, confirming their high inhibition behavior observed experimentally. Consequently, these results explain the adsorption of these inhibitors on the aluminum surface, forming a protective layer.

#### 4.2 Monte Carlo simulation result

Monte Carlo simulation is an effective technique to understand the adsorption behavior of corrosion inhibitors on the surface of a studied material.<sup>61</sup> Thus, we performed Monte Carlo simulations of the thiophene derivatives on an Al (111) surface in the presence of H<sub>2</sub>O. The most stable low-energy adsorption of the studied inhibitors molecules on the Al (111)/150H<sub>2</sub>O system is shown in Fig. 11.

As is known, corrosion inhibitors act by adsorption to protect materials from the corrosion environment.<sup>62</sup> Consequently, the adsorption energy values present a good method to classify the inhibitor efficiency.<sup>63</sup> Furthermore, the higher the negative adsorption energy value, the more stable the adsorption configuration of the studied molecules.<sup>64</sup> TET had a higher negative adsorption energy value compared to the OXM inhibitor (Table 6). This result explains the higher protective behavior of TET than the OXM inhibitor and the trend is the same as that obtained using the electrochemical and DFT methods. The two studied inhibitors adsorbed parallel on the aluminum surface to maximize the surface contact between the working electrode and studied molecules.<sup>65</sup> It is obvious that the adsorption energy values of TET and OXM are more negative compared to that of water molecules. This shows their ability to gradually substitute H<sub>2</sub>O solvent from the surface of the material, forming a stable film, which can protect the aluminum against corrosion in the aqueous phase.<sup>66–71</sup>

## Conclusion

The studied inhibitors showed a high inhibition performance to prevent the surface corrosion of the 2024-T3 aluminum alloy in contact with hydrochloric acid medium. This result was confirmed by diverse theoretical approaches and experimental techniques, besides the surface analysis characterization. It can be concluded from this study that:

- The EIS measurements indicate that thiophene products reached the maximum corrosion inhibition efficiencies of 94% and 91% for TET and OXM, respectively, at the optimum concentration of 10<sup>-3</sup> M, which decreased slightly with a decrease in their concentration.

- The Tafel curves indicated the mixed (cathodic and anodic)-type behavior of both the TET and OXM inhibitors.

- The adsorption behavior showed that studied inhibitors obeys the Al-Awady, Flory-Huggins and Temkin isotherm models. Therefore, a barrier film is formed on the aluminum surface, which was also justified by the surface characterization.

- The theoretical approach using DFT at the B3LYP level and Monte Carlo simulation results corroborate the experimental studies.

## Conflicts of interest

There are no conflicts to declare.

## Acknowledgements

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## References

- 1 Y. Liu, X. L. Li, J. F. Jin, J. A. Liu, Y. Y. Yan, Z. W. Han and L. Q. Ren, Anti-icing property of bioinspired micro-structure superhydrophobic surfaces and heat transfer model, *Appl. Surf. Sci.*, 2017, **400**, 498–505.
- 2 A. K. Rouniyar and P. Shandilya, Fabrication and experimental investigation of magnetic field assisted powder mixed electrical discharge machining on machining of aluminum 6061 alloy, *Proc. Inst. Mech. Eng. B J. Eng. Manuf.*, 2019, **233**, 2283–2291.
- 3 M. Becker, Chromate-free chemical conversion coatings for aluminum alloys, *Corros. Rev.*, 2019, **37**, 321–342.
- 4 M. Emadi, H. Beheshti, M. Heidari-Rarani and F. H. Aboutalebi, Experimental study of collapse mode and crashworthiness response of tempered and annealed aluminum tubes under axial compression, *J. Mech. Sci. Technol.*, 2019, **33**, 2067–2074.
- 5 Q. X. Kang, Y. Wang and X. Y. Zhang, Experimental and theoretical investigation on calcium oxide and L-aspartic as an effective hybrid inhibitor for aluminum-air batteries, *J. Alloys Compd.*, 2019, **774**, 1069–1080.
- 6 M. A. Deyab, Effect of nonionic surfactant as an electrolyte additive on the performance of aluminum-air battery, *J. Power Sources*, 2019, **412**, 520–526.
- 7 S. Q. Guo, J. J. Leavitt, X. Q. Zhou, Y. Xie, S. Tietze, Y. K. Zhu, A. Lawver, E. Lahti and J. S. Zhang, Effects of flow, Si inhibition, and concurrent corrosion of dissimilar metals on the corrosion of aluminium in the environment following a loss-of-coolant accident, *Corros. Sci.*, 2017, **128**, 100–109.
- 8 Q. S. Li, J. S. Wang and W. B. Hu, Optimizations of electric current assisted chemicalmilling condition of 2219 aluminum alloy, *J. Mater. Process. Technol.*, 2017, **249**, 379–385.
- 9 A. Boag, A. E. Hughes, N. C. Wilson, A. Torpy, C. M. Macrae, A. M. Glenn and T. H. Muster, How complex is the





- microstructure of AA2024-T3, *Corros. Sci.*, 2009, **51**, 1565–1568.
- 10 V. Guillaumin and G. Mankowski, Localised corrosion of 2024 T351 aluminum alloy in chloride media, *Corros. Sci.*, 1999, **41**, 421–438.
  - 11 C. Blanc, B. Lavallo and G. Mankowski, The role of precipitates enriched with copper on the susceptibility to pitting corrosion of the 2024 aluminium alloy, *Corros. Sci.*, 1997, **39**, 495–510.
  - 12 P. Campestrini, E. P. M. van Westing, H. W. van Rooijen and J. H. W. de Wit, Relation between microstructural aspects of AA2024 and its corrosion behaviour investigated using AFM scanning potential technique, *Corros. Sci.*, 2000, **42**, 1853–1861.
  - 13 T. Suter and R. C. Alkire, Microelectrochemical studies of pit initiation at single inclusions in Al 2024-T3, *J. Electrochem. Soc.*, 2001, **148**, 36–42.
  - 14 B. Elpern and F. C. Nachod, Absorption Spectra and Structure of Some Tetrazoles, *J. Am. Chem. Soc.*, 1950, **72**(8), 3379–3382.
  - 15 O. Schneider, G. O. Ilevbare, J. R. Scully and R. G. Kelly, In situ laser scanning microscopy of AA 2024-T3 corrosion metrology: II. Trench formation around particles, *J. Electrochem. Soc.*, 2004, **151**, 465–472.
  - 16 M. K. Cavanaugh, J.-C. Li, N. Birbilis and R. G. Buchheit, Electrochemical characterization of intermetallic phases common to aluminum alloys as a function of solution temperature, *J. Electrochem. Soc.*, 2014, **161**, 535–543.
  - 17 J. DeRose, T. Suter, A. Bałkowiec, J. Michalski, K. J. Kurzydłowski and P. Schmutz, Localized corrosion initiation and microstructural characterization of an Al2024 alloy with higher Cu to Mg ratio, *Corros. Sci.*, 2012, **55**, 313–325.
  - 18 W. Qafsaoui, M. W. Kendig, H. Perrot and H. Takenouti, Effect of 1-pyrrolidine on the galvanic coupling resistance of intermetallics—aluminium matrix during corrosion of AA 2024-T3 in dilute NaCl, *Corros. Sci.*, 2015, **92**, 245–255.
  - 19 K. H. Na and S. I. Pyun, Comparison of susceptibility to pitting corrosion of AA2024-T4, AA7075-T651 and AA7475-T761 aluminium alloys in neutral chloride solutions using electrochemical noise analysis, *Corros. Sci.*, 2008, **50**, 248–258.
  - 20 L. Domingues, J. C. S. Fernandes, M. D. Belo, M. G. S. Ferreira and L. Guerra-Rosa, Anodising of Al 2024-T3 in a modified sulphuric acid/boric acid bath for aeronautical applications, *Corros. Sci.*, 2003, **45**, 149–160.
  - 21 L. Pezzato, K. Brunelli and M. Dabalà, Corrosion properties of plasma electrolytic oxidation coated AA7075 treated using an electrolyte containing lanthanum-salts, *Surf. Interface Anal.*, 2016, **48**, 729–738.
  - 22 M. Mohedano, M. Serdechnova, M. Strykevich, S. Karpushenkov, A. C. Bouali, M. G. S. Ferreira and M. L. Zheludkevich, Active protective PEO coatings on AA2024: role of voltage on *in situ* LDH growth, *Mater. Des.*, 2017, **120**, 36–46.
  - 23 M. Costa and C. B. Klein, Toxicity and Carcinogenicity of Chromium Compounds in Humans, *Crit. Rev. Toxicol.*, 2006, **36**, 155–163.
  - 24 D. Snihirova, S. V. Lamaka, P. Taheri, J. M. C. Mol and M. F. Montemor, Comparison of the synergistic effects of inhibitor mixtures tailored for enhanced corrosion protection of bare and coated AA2024-T3, *Surf. Coat. Technol.*, 2016, **303**, 342–351.
  - 25 T. Zhang, W. Jiang, H. Wang and S. Zhang, Synthesis and localized inhibition behaviour of new triazine-methionine corrosion inhibitor in 1 M HCl for 2024-T3 aluminium alloy, *Mater. Chem. Phys.*, 2019, **237**, 121866.
  - 26 L. B. Coelho, D. Cossement and M. G. Olivier, Benzotriazole and cerium chloride as corrosion inhibitors for AA2024-T3: an EIS investigation supported by SVET and ToF-SIMS analysis, *Corros. Sci.*, 2018, **130**, 177–189.
  - 27 Y. Fernine, E. Ech-chihbi, N. Arrousse, F. El Hajjaji, F. Bousraf, M. EbnTouhami, Z. Rais and M. Taleb, Ocimum basilicum seeds extract as an environmentally friendly antioxidant and corrosion inhibitor for aluminium alloy 2024 -T3 corrosion in 3 wt% NaCl medium, *Colloids Surf. A Physicochem. Eng. Asp.*, 2021, **627**, 127232.
  - 28 S. A. Asipita, M. Ismail, M. Z. A. Majid, Z. A. Majid, C. Abdullah and J. Mirza, Green Bambusa Arundinacea leaves extract as a sustainable corrosion inhibitor in steel reinforced concrete, *J. Clean. Prod.*, 2014, **67**, 139–146.
  - 29 Y. Qiang, S. Zhang, L. Guo, S. Xu, L. Feng, I. B. Obot and S. Chen, Sodium dodecyl benzene sulfonate as a sustainable inhibitor for zinc corrosion in 26% NH<sub>4</sub>Cl solution, *J. Clean. Prod.*, 2017, **152**, 17–25.
  - 30 H. Bourzi, R. Oukhrib, B. El Ibrahimi, H. Abou Oualid, Y. Abdellaoui, B. Balkard and S. El Issami, Understanding of anti-corrosive behavior of some tetrazole derivatives in acidic medium: Adsorption on Cu (111) surface using quantum chemical calculations and Monte Carlo simulations, *Surf. Sci.*, 2020, **702**, 121692.
  - 31 K. Chkirate, K. Azgaou, H. Elmsellem, B. El Ibrahimi, N. K. Sebbar, M. Benmessaoud and E. M. Essassi, Corrosion inhibition potential of 2-[(5-methylpyrazol-3-yl)methyl] benzimidazole against carbon steel corrosion in 1 M HCl solution: Combining experimental and theoretical studies, *J. Mol. Liq.*, 2021, **321**, 114750.
  - 32 A. S. Fouda and H. A. Wahed, Corrosion inhibition of copper in HNO<sub>3</sub> solution using thiophene and its derivatives, *Arab. J. Chem.*, 2016, **9**, S91–S99.
  - 33 A. K. Singh, S. Thakur, B. Pani, B. Chugh, H. L. Gaz, I.-M. Chung, P. Chaubey, A. K. Pandey and J. Singh, A Solvent-free microwave assisted synthesis and corrosion inhibition study of a series of hydrazones derived from thiophene derivatives: experimental, surface and theoretical study, *J. Mol. Liq.*, 2019, **283**, 788–803.
  - 34 L. J. Zhang and G. Zhang Tao, A novel and high-efficiency inhibitor of 5-(4-methoxyphenyl)-3H-1,2-dithiole-3-thione for copper corrosion inhibition in sulfuric acid at different temperatures, *J. Mol. Liq.*, 2018, **272**, 369–379, DOI: 10.1016/j.molliq.2018.09.095.



- 35 Y. J. Qiang, S. T. Zhang, L. Guo, X. W. Zheng, B. Xiang and S. J. Chen, Experimental and theoretical studies of four allyl imidazolium-based ionic liquids as green inhibitors for copper corrosion in sulfuric acid, *Corros. Sci.*, 2017, **199**, 68–78, DOI: 10.1016/j.corsci.2017.02.021.
- 36 M. K. Awad, “Quantum chemical studies and molecular modelling of the effect of polyethylene glycol as corrosion inhibitors of an aluminium surface”, *Can. J. Chem.*, 2012, **91**, 283–291.
- 37 B. G. Prakashaiaha, A. Nityananda Shetty and B. E. Amitharana, 2-(4-(Diethylamino)-2-hydroxybenzylidene) Hydrazinecarboamide as Corrosion Inhibitor on AA2024-T3 Aluminium Alloy in 0.5 M Hydrochloric Acid Solution, *Surf. Eng. Appl. Electrochem.*, 2018, **5**, 286–296.
- 38 W. Luo, W. Li, J. Tan, J. Liu, B. Tan, X. Zuo, Z. Wang and X. Zhang, Saccharum Officinarum Leaf Extract as Corrosion Inhibitor of Copper Corrosion in Sulphuric Acid Solution: Experiments and Theoretical Calculations, *J. Mol. Liq.*, 2020, **314**, 113630.
- 39 H. Gerengi, M. Mielniczek, G. Gece and M. M. Solomon, Experimental and quantum chemical evaluation of 8-hydroxyquinoline as a corrosion inhibitor for copper in 0.1 M HCl, *Ind. Eng. Chem. Res.*, 2016, **55**, 9614–9624.
- 40 A. Y. S. E. L. Yurt, S. Ulutas and H. Dal, “Electrochemical and theoretical investigation on the corrosion of aluminum in acidic solution containing some Schiff bases”, *Appl. Surf. Sci.*, 2006, **253**(2), 919–925.
- 41 B. Hirschorn, M. E. Orazem, B. Tribollet, V. Vivier, I. Frateur and M. Musiani, Determination of effective capacitance and film thickness from constant-phase-element parameters, *Electrochim. Acta*, 2010, **55**, 6218–6227.
- 42 T. Zhang, W. Jiang, H. Wang and S. Zhang, Synthesis and localized inhibition behaviour of new triazine-methionine corrosion inhibitor in 1 M HCl for 2024-T3 aluminum alloy, *Mater. Chem. Phys.*, 2019, **237**, 121866.
- 43 A. Saady, Z. Rais, F. Benhiba, R. Salim, K. Ismaili Alaoui, N. Arrousse, F. El hajjaji, M. Taleb, K. Jarmoni, Y. Kandri Rodi, I. Warad and A. Zarrouk, Chemical, electrochemical, quantum, and surface analysis evaluation on the inhibition performance of novel imidazo[4,5-*b*] pyridine derivatives against mild steel corrosion, *Corros. Sci.*, 2021, **189**, 109621, DOI: 10.1016/j.corsci.2021.109621.
- 44 N. Arrousse, R. Salim, F. Benhiba, E. H. Mabrouk, A. Abdelaoui, F. El-Hajjaji, I. Warad, A. Zarrouk and M. Taleb, Insight into the corrosion inhibition property of two new soluble and non-toxic xanthenbenzoate derivatives, *J. Mol. Liq.*, 2021, **338**, 116610, DOI: 10.1016/j.molliq.2021.116610.
- 45 E. X. Ricky, M. pelwa and M. Xu, The study of *m*-pentadecylphenol on the inhibition of mild steel corrosion in 1 M HCl solution, *J. Ind. Eng. Chem.*, 2021, **101**, 359–371, DOI: 10.1016/j.jiec.2021.05.047.
- 46 G. Catapano, F. G. Sgulo, V. Seneca, G. Lepore, L. Columbano and G. di Nuzzo, Fluorescein-Guided Surgery for High-Grade Glioma Resection: An Intraoperative “Contrast-Enhancer”, *World Neurosurg.*, 2017, **104**, 239–247, DOI: 10.1016/j.wneu.2017.05.022.
- 47 J. P. Ehlers, K. Wang, A. Vasanji, M. Hu and S. K. Srivastava, Automated quantitative characterisation of retinal vascular leakage and microaneurysms in ultra-widefield fluorescein angiography, *Br. J. Ophthalmol.*, 2017, **101**, 696–699, DOI: 10.1136/bjophthalmol-2016-310047.
- 48 J. Sathiyabama, S. Rajendran, A. Selvi and J. Jeyasundari, Fluorescein as corrosion inhibitor for carbon steel in well water, *Bulg. Chem. Commun.*, 2009, **41**, 374–379.
- 49 E. E. Ebenso and E. E. Oguzie, Corrosion inhibition of mild steel in acidic media by some organic dyes, *Mater. Lett.*, 2005, **59**, 2163–2165, DOI: 10.1016/j.matlet.2005.02.055.
- 50 N. Arrousse, R. Salim, A. Abdellaoui, F. El hajjaji, B. Hammouti, A. D. Wilson, E. Mabrouk and M. Taleb, Synthesis. characterization. and evaluation of xanthene derivative as highly effective. nontoxic corrosion inhibitor for mild steel immersed in 1 M HCl solution, *J. Taiwan Inst. Chem. Eng.*, 2021, **120**, 344–359, DOI: 10.1016/j.jtice.2021.03.026.
- 51 S. Alaoui Mrani, E. Ech-chihbi, N. Arrousse, Z. Rais, F. El Hajjaji, C. El Abiad, S. Radi, J. Mabrouki, M. Taleb and S. Jodeh, DFT and Electrochemical Investigations on the Corrosion Inhibition of Mild Steel by Novel Schiff's Base Derivatives in 1 M HCl Solution, *Arab. J. Sci. Eng.*, 2021, **46**, 5691–6570, DOI: 10.1007/s13369-020-05229-4.
- 52 E. K. Ardakani and E. Kowsari, A. Ehsani Imidazolium-derived polymeric ionic liquid as a green inhibitor for corrosion inhibition of mild steel in 1.0 M HCl: Experimental and computational study, *Colloids Surf. A Physicochem. Eng. Asp.*, 2020, **586**, 124195, DOI: 10.1016/j.colsurfa.2019.124195.
- 53 A. Zarrouk, B. Hammouti, T. Lakhli and F. Bentiss, New 1-H-pyrrole-2,5-dione derivatives as efficient organic inhibitors of carbon steel corrosion in hydrochloric acid medium: Electrochemical. XPS and DFT studies, *Corros. Sci.*, 2015, **90**, 572–584, DOI: 10.1016/j.corsci.2014.10.052.
- 54 R. G. Pearson, Absolute Electronegativity and Hardness: Application to Inorganic Chemistry, *Inorg. Chem.*, 1988, **27**, 734–740, DOI: 10.1021/ic00277a030.
- 55 I. B. Obot, K. Haruna and T. A. Saleh, Atomistic Simulation: A Unique and Powerful Computational Tool for Corrosion Inhibition Research, *Arab. J. Sci. Eng.*, 2019, **44**, 1–32, DOI: 10.1007/s13369-018-3605-4.
- 56 R. G. Parr and R. G. Pearson, Absolute hardness: companion parameter to absolute electronegativity, *J. Am. Chem. Soc.*, 1983, **105**, 7512–7516, DOI: 10.1021/ja00364a005.
- 57 R. G. Pearson, Absolute electronegativity and hardness correlated with molecular orbital theory, *Proc. Natl. Acad. Sci. U.S.A.*, 1986, **83**, 8440–8441, DOI: 10.1073/pnas.83.22.8440.
- 58 N. Arrousse, El. Mabrouk, K. Ismaili Alaoui, F. El hajjaji, Z. Rais and M. Taleb, Economical and new synthesis strategy, characterization and theoretical study of 3-oxo-3H-spiro [isobenzofuran-1,9'-xanthene]-3'.6'-diyl dibenzoate, *SN Appl. Sci.*, 2020, **2**(5), 1019, DOI: 10.1007/s42452-020-2782-4.



- 59 H. Ouici, M. Tourabi, O. Benali, C. Selles, C. Jama, A. Zarrouk and F. Bentiss, Adsorption and corrosion inhibition properties of 5-amino 1, 3, 4-thiadiazole-2-thiol on the mild steel in hydrochloric acid medium: Thermodynamic, surface and electrochemical studies, *J. Electroanal. Chem.*, 2017, **803**, 25–34, DOI: 10.1016/j.jelechem.2017.09.018.
- 60 D. K. Singh, S. Kumar, G. Udayabhanu and R. P. John, 4 (N-N-dimethylamino) benzaldehyde nicotinic hydrazone as corrosion inhibitor for mild steel in 1 M HCl solution: An experimental and theoretical study, *J. Mol. Liq.*, 2016, **216**, 738–746, DOI: 10.1016/j.molliq.2016.02.012.
- 61 N. Arrousse, R. Salim, Y. Kaddouri, A. Zarrouk, D. Zahri, F. El Hajjaji, R. Touzani, M. Taleb and S. Jodeh, The inhibition behavior of Two pyrimidine-pyrazole derivatives against corrosion in hydrochloric solution: experimental. surface analysis and *in silico* approach studies, *Arab. J. Chem.*, 2020, **13**, 5949–5965, DOI: 10.1016/j.arabjc.2020.04.030.
- 62 N. Arrousse, R. Salim, G. Al Houari, F. El hajjaji, A. Zarrouk, Z. Rais, M. Taleb, D. S. Chauhan and M. A. Quraishi, Experimental and theoretical insights on the adsorption and inhibition mechanism of (2*E*)-2-(acetylamino)-3-(4-nitrophenyl) prop-2-enoic acid and 4-nitrobenzaldehyde on mild steel corrosion, *J. Chem. Sci.*, 2020, **132**, 112, DOI: 10.1007/s12039-020-01818-w.
- 63 N. Arrousse, E. Mabrouk, B. Hammouti, F. El hajjaji, Z. Rais and M. Taleb, Synthesis, characterization, anti-corrosion behavior and theoretical study of the new organic dye: 3-oxo-3*H* spiro[isobenzofuran-1.9'-xanthene]-3', 6'-diyl bis(3-methylbenzenesulfonate), *Int. J. Corros. Scale Inhib.*, 2020, **9**, 661–687, DOI: 10.17675/2305-6894-2020-9-2-18.
- 64 Z. Shariatnia and A. Ahmadi-Ashtiani, Corrosion inhibition efficiency of some phosphoramidate derivatives: DFT computations and MD simulations, *J. Mol. Liq.*, 2019, **298**, 111–409, DOI: 10.1016/j.molliq.2019.111409.
- 65 J. Lazrak, R. Salim, N. Arrousse, E. Ech-chihbi, F. El-Hajjaji, M. Taleb, A. Farah and A. Ramzi, Mentha viridis oil as a green effective corrosion inhibitor for mild steel in 1 M HCl medium, *Int. J. Corros. Scale Inhib.*, 2020, **9**, 1580–1606, DOI: 10.17675/2305-6894-2020-9-4-25.
- 66 O. Abdellaoui, M. K. Skalli, A. Haoudi, Y. Kandri Rodi, N. Arrousse, M. Taleb, R. Ghibate and O. Senhaji, Study of the inhibition of corrosion of mild steel in a 1 M HCl solution by a new quaternary ammonium surfactant, *Mor. J. Chem.*, 2021, **9**, 044–056, DOI: 10.48317/IMIST.PRSM/morjchem-v9i1.21313.
- 67 N. Arrousse, E. Mabrouk, R. Salim, K. Ismaili alaoui, F. El Hajjaji, Z. Rais, M. Taleb and B. Hammouti, Fluorescein as commercial and environmentally friendly inhibitor against corrosion of mild steel in molar hydrochloric acid medium, *Mater. Today: Proc.*, 2020, **27**, 3184–3192, DOI: 10.1016/j.matpr.2020.04.201.
- 68 N. Arrousse, A. Nahlé, E. Mabrouk, R. Salim, F. El hajjaji, Z. Rais and M. Taleb, An Intelligent and Efficient Synthesis of New Inhibitors against Corrosion of Mild Steel in Acidic Media, *Surf. Eng. Appl. Electrochem.*, 2021, **57**(1), 136–147, DOI: 10.1016/j.molliq.2021.116610.
- 69 N. M. El Basionya, E. H. Tawfik, M. A. El-raouf, A. A. Fadda and M. M. Waly, Synthesis, characterization, theoretical calculations (DFT and MC), and experimental of different substituted pyridine derivatives as corrosion mitigation for X-65 steel corrosion in 1 M HCl, *J. Mol. Struct.*, 2021, **1231**, 129999, DOI: 10.1016/j.molstruc.2021.129999.
- 70 H. Zarrok, A. Zarrouk, R. Salghi, H. Oudda, B. Hammouti, M. Ebn Touhami, M. Bouachrine and S. Boukhris, A Combined Experimental and Theoretical Study on the Corrosion Inhibition and Adsorption Behaviour of Quinoxaline Derivative during Carbon Steel Corrosion in Hydrochloric Acid, *Port. Electrochimica Acta*, 2012, **30**, 405–417, DOI: 10.4152/pea.201206405.
- 71 F. EL Hajjaji, R. Salim, M. Taleb, F. Benhiba, N. Rezki, D. S. Chauhan and M. A. Quraishi, Pyridinium-based ionic liquids as novel eco-friendly corrosion inhibitors for mild steel in molar hydrochloric acid: Experimental & computational approach, *Surf. Interfaces*, 2021, **22**, 100881, DOI: 10.1016/j.surfin.2020.100881.

