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# Metal oxide nanoparticle-modified ITO electrode for high-performance solution-processed perovskite photodetectors

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Low dark current density plays a key role in determining the overall performance of perovskite photodetectors (PPDs). To achieve this goal, a hole transport layer (HTL) on the ITO side and a hole blocking layer (HBL) on the metal electrode side are commonly introduced in PPDs. Unlike traditional approaches, we realized a high-performance solution-processed broadband PPD using metal oxide (MO) nanoparticles (NPs) as the HBL on the ITO electrode and PC<sub>61</sub>BM as another HBL on the metal electrode side to reduce the device dark current. The PPDs based on TiO<sub>2</sub> and SnO<sub>2</sub> NP-modified layers show similar device performances at  $-0.5$  V: a greater than  $10^5$  on/off ratio; over 100 dB linear dynamic range (LDR) under different visible light illumination; around  $0.2$  A W<sup>-1</sup> responsivity ( $R$ ); greater than  $10^{12}$  jones detectivity ( $D^*$ ); and  $\sim 20$   $\mu$ s rise time of the device. The MO NP interfacial layer can significantly suppress charge injection in the dark, while the accumulated photogenerated charges at the interface between the MO layer and the perovskite layer introduce band bending, leading to dramatically increased current under illumination. Therefore, the dark current density of the devices is significantly reduced and the optical gain is drastically enhanced. However, after UV illumination, the dark current of the TiO<sub>2</sub> device dramatically increases while the dark current of the SnO<sub>2</sub> device can stay the same as before since the UV illumination-induced conductivity and barrier height changes in the TiO<sub>2</sub> layer cannot recover after removing the UV irradiation. These results indicate that the TiO<sub>2</sub> NP layer is suitable for making a vis-NIR photodetector, while the SnO<sub>2</sub> NP layer is a good candidate for UV-vis-NIR photodetectors. The facile solution-processed high-performance perovskite photodetector using MO NP-modified ITO is highly compatible with low cost, flexible, and large-area electronics.

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## Introduction

Ever since solid organic–inorganic hybrid perovskites have been applied in solar cells, their power conversion efficiency (PCE) has rapidly risen from 3.8%<sup>1</sup> to over 25%.<sup>2</sup> In addition to the significant progress seen in solar cells, perovskite materials have made a series of breakthroughs in light-emitting diodes (LEDs),<sup>3–5</sup> amplified spontaneous emission (ASE) or lasers<sup>6,7</sup> and photodetectors (PDs).<sup>8,9</sup> Compared with traditional photodetectors based on Si,<sup>10</sup> GaN,<sup>11,12</sup> and GaAs,<sup>13,14</sup> perovskite-based photodetectors have attracted much attention due to their unique advantages of low temperature solution process, tunable bandgap, high quantum efficiency, broad wavelength response, and large area application. However, the mass surface defects and grain boundaries result in solution-processed perovskite PDs usually suffering from a high dark current due to the trapped charge carrier-induced leakage current or

enhanced charge injection under reverse bias.<sup>15</sup> The dark current ( $J_d$ ) plays a crucial role in determining the overall performance of PDs, impacting the signal-to-noise ratio (SNR), detectivity ( $D^*$ ), and linear dynamic range (LDR). Therefore, in common PPDs, a thin phenyl-C61-butyric acid methyl ester (PC<sub>61</sub>BM) film is usually deposited on top of the perovskites to passivate the defects.<sup>16</sup> For superior device performance, various hole-blocking layers (HBL) have been added between the perovskite layer and the metal electrode to suppress the leakage current. Yang *et al.* introduced PEDOT as a hole transport layer (HTL) and poly[(9,9-bis(30-(*N,N*-dimethylamino)propyl)-2,7-fluorene)-*alt*-2,7-(9,9-dioctylfluorene)] (PFN) or 2,9-dimethyl-4,7-diphenyl-1,10-phenanthroline (BCP) as HBL to achieve a high on/off ratio of  $10^5$ ;<sup>17</sup> Meredith *et al.* applied PEDOT and C<sub>60</sub> on top of the ITO layer and PC<sub>61</sub>BM layer, respectively, to achieve low noise;<sup>18</sup> Huang *et al.* coated the C<sub>60</sub>/BCP bilayer between PC<sub>61</sub>BM and Al to realize a low noise perovskite photodetector;<sup>19</sup> Huang *et al.* used poly[bis(4-phenyl)(2,4,6-trimethylphenyl)amine] (PTAA) as HTL and IEICO/C<sub>60</sub>/BCP trilayer on top of the perovskite layer to demonstrate a highly sensitive photodetector;<sup>20</sup> Yu *et al.* used

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4,4'-cyclohexylidenebis[*N,N*-bis(*p*-tolyl) aniline] (TAPC) as HTL and introduced the PBDB/IHIC bulk heterojunction between the perovskite and PCBM to reduce the undesired high dark current;<sup>21</sup> Chen *et al.* proposed a self-assembled polar monolayer (SAPM) as HTL and used PC<sub>61</sub>BM/PEI as HBL to achieve high performance; Alshehri *et al.*,<sup>22</sup> Caironi *et al.*,<sup>23</sup> Oh *et al.*,<sup>24</sup> and Ruan *et al.*<sup>25</sup> have demonstrated a series of high-performance perovskite photodetectors by introducing PDPP3T:PC<sub>61</sub>BM, PC<sub>61</sub>BM/AZO, PC<sub>61</sub>BM/TiO<sub>x</sub>, and PC<sub>61</sub>BM/PCBM:F4-TCNQ/BCP as the hole-blocking layer in the PEDOT-modified ITO-based perovskite photodetectors, respectively. The abovementioned works show that a typical high-performance PPD typically needs HTL on the ITO side and HBL on the metal electrode side. In these devices, PEDOT is usually used as HTL and the mixed layer or multi-layer containing fullerene is used as HBL. The poly(triaryl amine) (PTAA)-modified atomic layer deposition (ALD) ZnO film has also been explored to make high-performance perovskite photodetectors with a configuration of ITO/ZnO/PTAA/perovskite/PC<sub>61</sub>BM/Ag.<sup>26</sup> It should be mentioned that, unlike TiO<sub>2</sub> or SnO<sub>2</sub>, when the annealing temperature is greater than 100 °C, ZnO can react with perovskite and affect the film formation and stability of the perovskite.<sup>27–29</sup> Therefore, in order to get a better perovskite film on ZnO, it is usually necessary to add a modified layer on ZnO or anneal ZnO at high temperature in advance to form a more compact ZnO film. Different from previous studies, we propose and demonstrate a high-performance PPD using MO NP HBL on the ITO side and PCBM HBL on the Al side. Either in-house synthesized TiO<sub>2</sub> NPs<sup>30</sup> or commercial SnO<sub>2</sub><sup>31</sup> were coated on ITO, as in our previously reported method, and both resulting PPDs show comparably low  $J_d$  of  $\sim 40$  nA cm<sup>-2</sup> at  $-0.5$  V, leading to greater than 10<sup>5</sup> SNR, more than 10<sup>12</sup> Jones  $D^*$ , and over 100 dB LDR under different light illuminations. Moreover, the comparison of UV illumination effects on the dark current of the TiO<sub>2</sub> and SnO<sub>2</sub> devices reveals that the former is suitable for vis-NIR photodetector application, while the latter is good for UV-vis-NIR photodetector application. This novel device structure with the single hole blocking layer on both ITO and Al sides not only significantly reduces the dark current of the device, but also greatly simplifies the device fabrication process, paving a new way for high-performance perovskite photodetector fabrications.

## Results and discussion

The energy diagrams of the devices are shown in the insets of Fig. 1a and b, respectively. The thin TiO<sub>2</sub> or SnO<sub>2</sub> NP layer was employed on top of the ITO as HBL. The TiO<sub>2</sub> nanoparticles and SnO<sub>2</sub> nanoparticles have similar conduction band minimum (CBM) and valence band maximum (VBM), and they are  $-3.9$  eV/ $-3.8$  eV and  $-7.2$  eV/ $-7.6$  eV, respectively. The MAPbI<sub>3</sub> layer was used as a light-absorber and the PC<sub>61</sub>BM film was spin-coated on the top of the perovskite layer as the top HBL. The  $J$ - $V$  curves of the TiO<sub>2</sub> and SnO<sub>2</sub> devices were measured under the bias from 1 V to  $-1$  V, as shown in Fig. 1a and b. In the dark, the low current densities are  $3.81 \times 10^{-8}$  A cm<sup>-2</sup> for the TiO<sub>2</sub> device and  $4.43 \times 10^{-8}$  A cm<sup>-2</sup> for the SnO<sub>2</sub> device at  $-0.5$  V,

respectively, indicating that the charge injection is significantly suppressed at both cathode and anode interfaces. The current rectification implies that the electron injection from the Al electrode through the whole device is obviously easier than that from the ITO electrode because of the higher barrier height between ITO and the TiO<sub>2</sub> or SnO<sub>2</sub> layer. Meanwhile, under standard AM 1.5 G sunlight illumination, the current density of both devices increase to  $10^{-3}$  A cm<sup>-2</sup> ( $9.03 \times 10^{-3}$  A cm<sup>-2</sup> for the TiO<sub>2</sub> device and  $7.39 \times 10^{-3}$  A cm<sup>-2</sup> for the SnO<sub>2</sub> device at  $-0.5$  V). Therefore, over 10<sup>5</sup> on/off ratio ( $2.36 \times 10^5$  at  $-0.5$  V for the TiO<sub>2</sub> device,  $1.67 \times 10^5$  at  $-0.5$  V) can be achieved, comparable to previously reported results.<sup>17,20,21,23,24,32</sup>

The external quantum efficiency (EQE) is an important parameter that characterizes the ratio between the number of photo-induced charge carriers collected by the external circuit and the number of incoming photons. The EQE of the devices was measured at 0 V and  $-0.5$  V, and the results are shown in Fig. 1c, d. The scanning wavelength is from 300 to 850 nm, and the maximum EQE values of the TiO<sub>2</sub> device and the SnO<sub>2</sub> device at 0 V are 33% and 51%, respectively. At  $-0.5$  V, the maximum EQE values for the TiO<sub>2</sub> device and the SnO<sub>2</sub> device increase to 73.4% at 530 nm and 60.1% at 670 nm, respectively. Responsivity ( $R$ ) is the ratio of output current to input optical power, which can be calculated from EQE using the following eqn (1):

$$R = EQEql/h\nu \quad (1)$$

where  $h\nu$  is the energy of the incident photon and  $q$  is the elementary charge of the electron. The ability to detect the optical signals from the noise is usually measured by detectivity, which can be calculated by the eqn (2):

$$D^* = R / \sqrt{2qJ_{\text{dark}}} \quad (2)$$

where  $J_{\text{dark}}$  is the dark current density. The corresponding  $R$  and  $D^*$  of the TiO<sub>2</sub> device and the SnO<sub>2</sub> device are shown in Fig. 1e and f. The maximum  $R$  is around  $0.32$  A W<sup>-1</sup> ( $-0.5$  V) for the TiO<sub>2</sub> device, and it is  $0.26$  A W<sup>-1</sup> for the SnO<sub>2</sub> device ( $-0.5$  V). The  $D^*$  of the TiO<sub>2</sub> device (Fig. 1g) reaches  $2.17 \times 10^{12}$  (0 V, 550 nm) while that of the SnO<sub>2</sub> device (Fig. 1h) is  $1.52 \times 10^{12}$  (0 V, 550 nm). It is worth noting that the detectivity decreases when the bias increases from 0 V to  $-0.5$  V, attributed to the increase of the dark current.

Response time is another critical parameter for the practical application of the photodetector. It includes the rise time and fall time: the former is defined as the time required for the current value to increase from 10% to 90% of the maximum value, and the latter is the current value falling from 90% to 10% of the peak value. Rise time reflects the process of generation, diversion, transportation, and collection of the photo-induced carriers, while fall time is dependent on the release of the trapped carriers by defects when the light source is removed. The response times of the two devices were measured under the illumination of 530 nm light at  $-0.5$  V, and the results are shown in Fig. 2a and b. The rise time for the TiO<sub>2</sub> and SnO<sub>2</sub> devices are 19.08  $\mu$ s and 24.83  $\mu$ s, respectively. The fall time of the SnO<sub>2</sub> device is 134.7  $\mu$ s



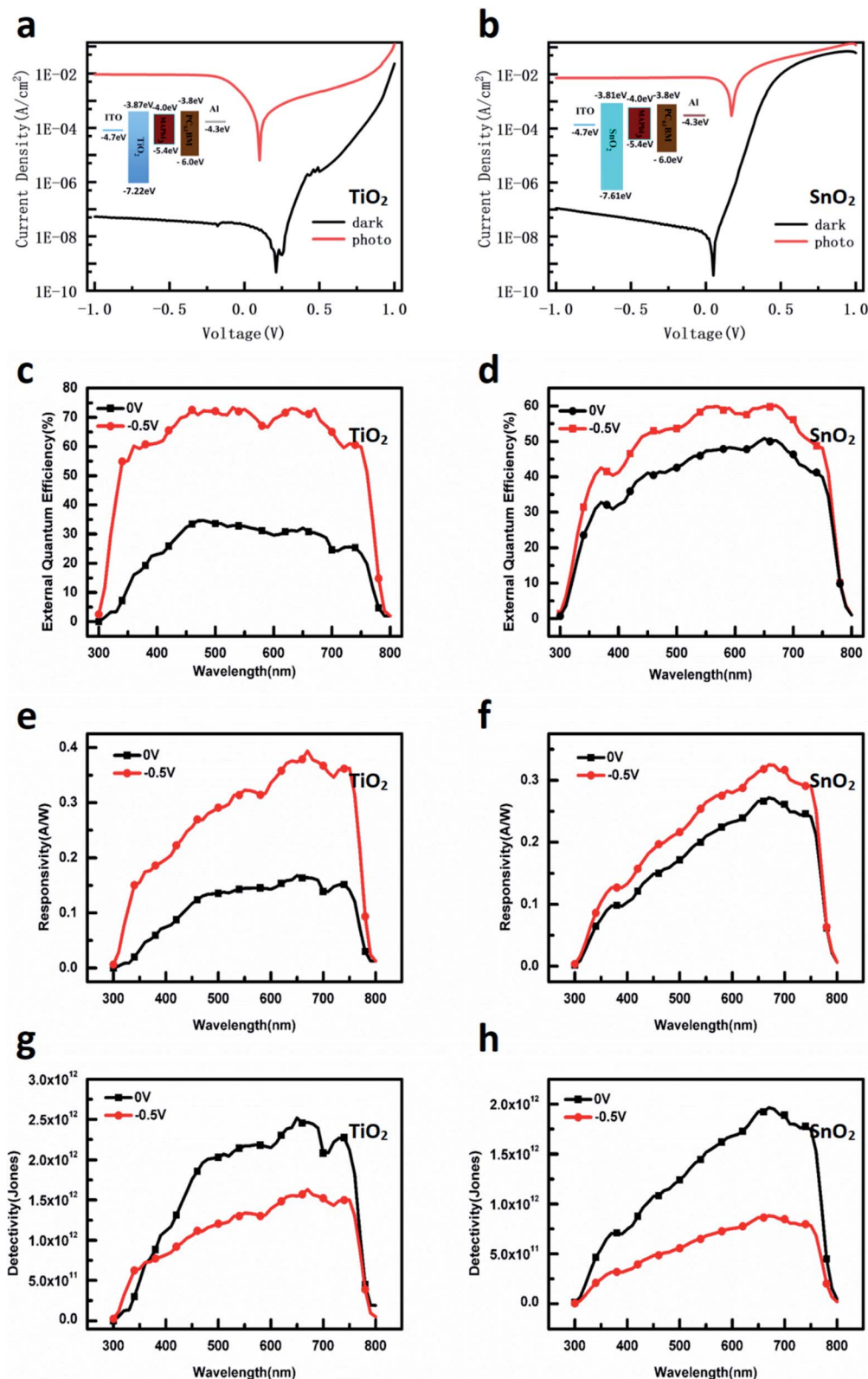


Fig. 1 The  $J$ - $V$  characteristics of the devices in the dark or under illumination, and the schematic energy level diagram of the devices: (a) the device based on TiO<sub>2</sub> nanoparticles, (b) the device based on SnO<sub>2</sub> nanoparticles; the photoresponse of the devices: (c, e and g) the EQE,  $R$ , and  $D^*$  of the TiO<sub>2</sub> device, respectively; (d, f and h) the EQE,  $R$ , and  $D^*$  of the SnO<sub>2</sub> device, respectively.



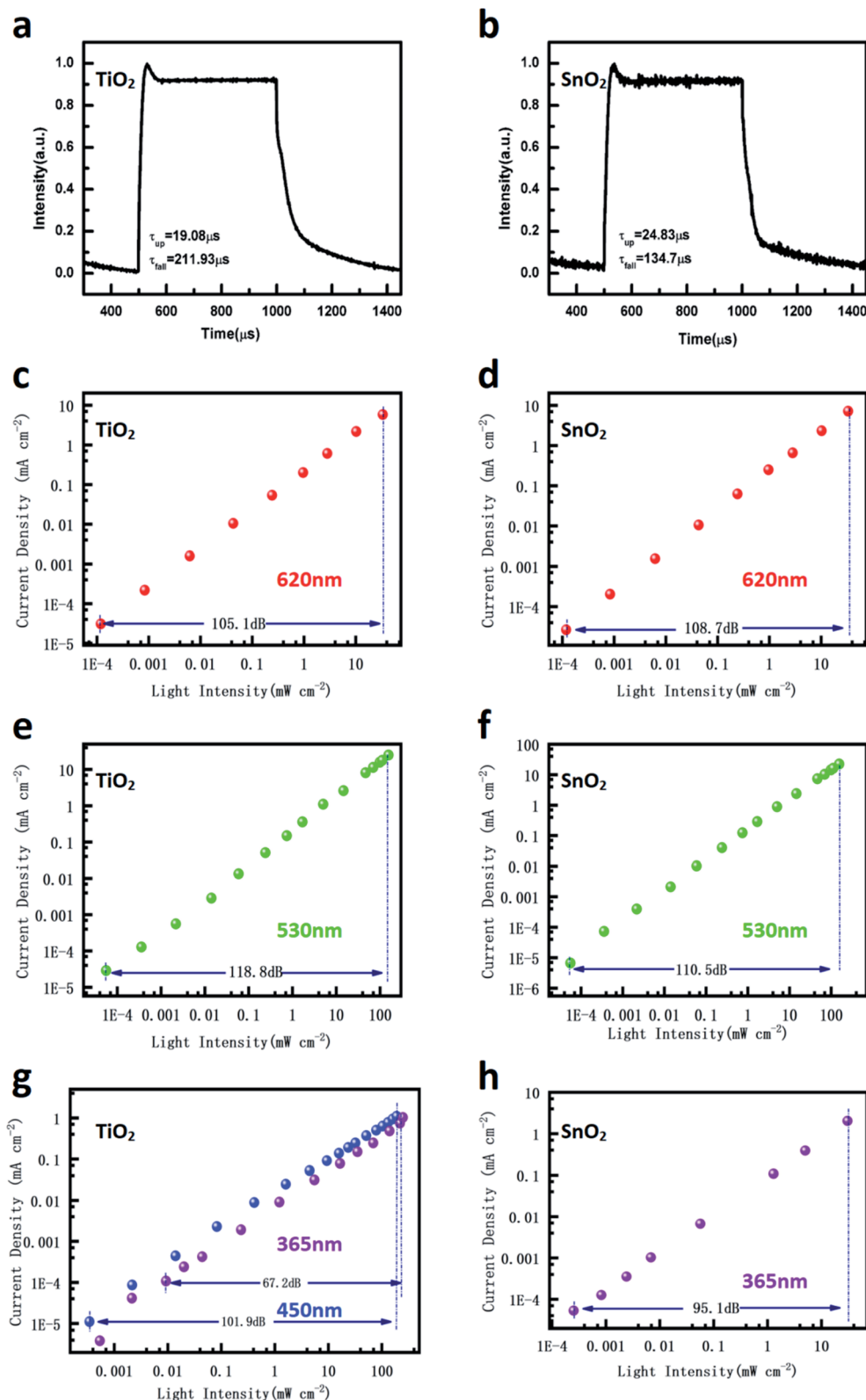


Fig. 2 Response time of (a) the  $\text{TiO}_2$  device; (b) the  $\text{SnO}_2$  device; LDRs of the  $\text{TiO}_2$  and  $\text{SnO}_2$  devices measured at  $-0.5$  V under various wavelength illuminations (c–h).

and the  $\text{TiO}_2$  device is  $211.93 \mu\text{s}$ , indicating that the trapped charges in the  $\text{TiO}_2$  NP layer need more release time than those in the  $\text{SnO}_2$  NP layer. From the practical perspectives, LDR is very

crucial for the photodetector. LDR is defined as the range over which the photocurrent is linear with light intensity. Beyond this range, the relationship between the light signal and electrical



signal is not linear anymore and the light signal cannot be calculated precisely. The devices were tested under the illuminations of 530 nm and 620 nm LEDs, respectively, and the results are shown in Fig. 2c–f. Both devices show a large linear range under different incident light intensities (power density ranges from  $10^{-1}$  to  $10^{-7}$  W cm $^{-2}$ ), leading to a LDR of over 100 dB. However, when the devices were tested under the illumination of UV light (Fig. 2g and h), the LDR of TiO $_2$  is significantly dropped to 65 dB while the SnO $_2$  still has 95 dB value. When using 450 nm light, the LDR of the TiO $_2$  device changes back to 100 dB, implying the significantly different sensitivities between the TiO $_2$  and SnO $_2$  NP layer to UV light.

The wavelength (with or without UV)-dependent  $J$ - $V$  curves were measured and the results are shown in Fig. 3. As Fig. 3a and b show, compared with the sunlight illumination, when a 400 nm filter is used, the photocurrents under bias for both devices are slightly reduced. In contrast, the open voltage of the SnO $_2$  device does not show obvious changes under different wavelength illuminations, while the open voltage of the TiO $_2$  device increases by about 0.3 V compared with the added UV illumination. Considering the same structure and measurement condition, the result is attributed to the different UV sensitivities to the TiO $_2$  NP layer compared with the SnO $_2$  NP layer. The dark currents of the two devices were measured after different light exposures for 1 min, as shown in Fig. 3c and d. When the TiO $_2$  device was exposed under illumination without UV wavelength, the device dark current was retested and the value is almost the same as the one before illumination. However, after illumination with the full wavelength, the dark

current increases to a high level, which illustrates that the device is highly conductive. The SnO $_2$  device shows consistently low dark current before and after illumination with or without UV wavelength. These behaviors indicate that the short-term UV illumination induces longtime effects in the TiO $_2$  device, but not in the SnO $_2$  device. It has been widely reported that TiO $_2$  has photocatalysis property under UV light and SnO $_2$  is not sensitive to UV light.<sup>33,34</sup> Therefore, the results suggest that the UV illumination-induced carriers may store in the TiO $_2$  NP layer, which would influence its conductivity and the energy level, resulting in a high dark current when removing the illumination and the different output property with or without UV light.

As shown in Fig. 4a and b, different wavelength filters are used to distinguish the influence of various wavelength lights for the TiO $_2$  device. The  $V_{oc}$  gradually decreases from 0.48 V to 0.13 V when the filters from 370 nm to 330 nm are used. Compared with the  $V_{oc}$  of the device under full wavelength illumination, the value used 290 nm filter does not obviously change. This result suggests that with the shorter wavelength light absorbed by the TiO $_2$  NPs, there is a larger Fermi energy level change in the TiO $_2$  NP layer. Fig. 4c and d show the dark currents and the light currents under illuminations with a 400 nm filter of the devices after different light exposure conditions, as shown in Fig. 4a. Similar to the above observations, shortening the illumination wavelength can significantly increase the dark current of the device. Moreover, the photocurrent (Fig. 4d) shows that the UV illumination-induced  $V_{oc}$  change shown in Fig. 4a cannot recover to the maximum value, indicating that the energy level change remained.

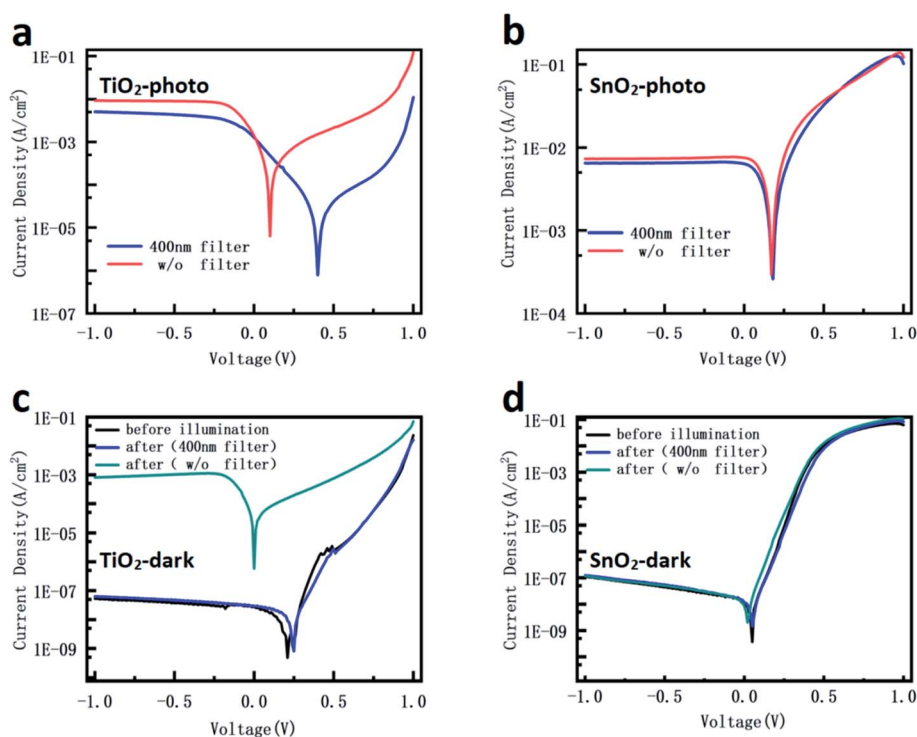


Fig. 3 The wavelength-dependent  $J$ - $V$  curves of the TiO $_2$  device: (a) under illumination, (c) in the dark; and the SnO $_2$  device: (b) under illumination, (d) in the dark.



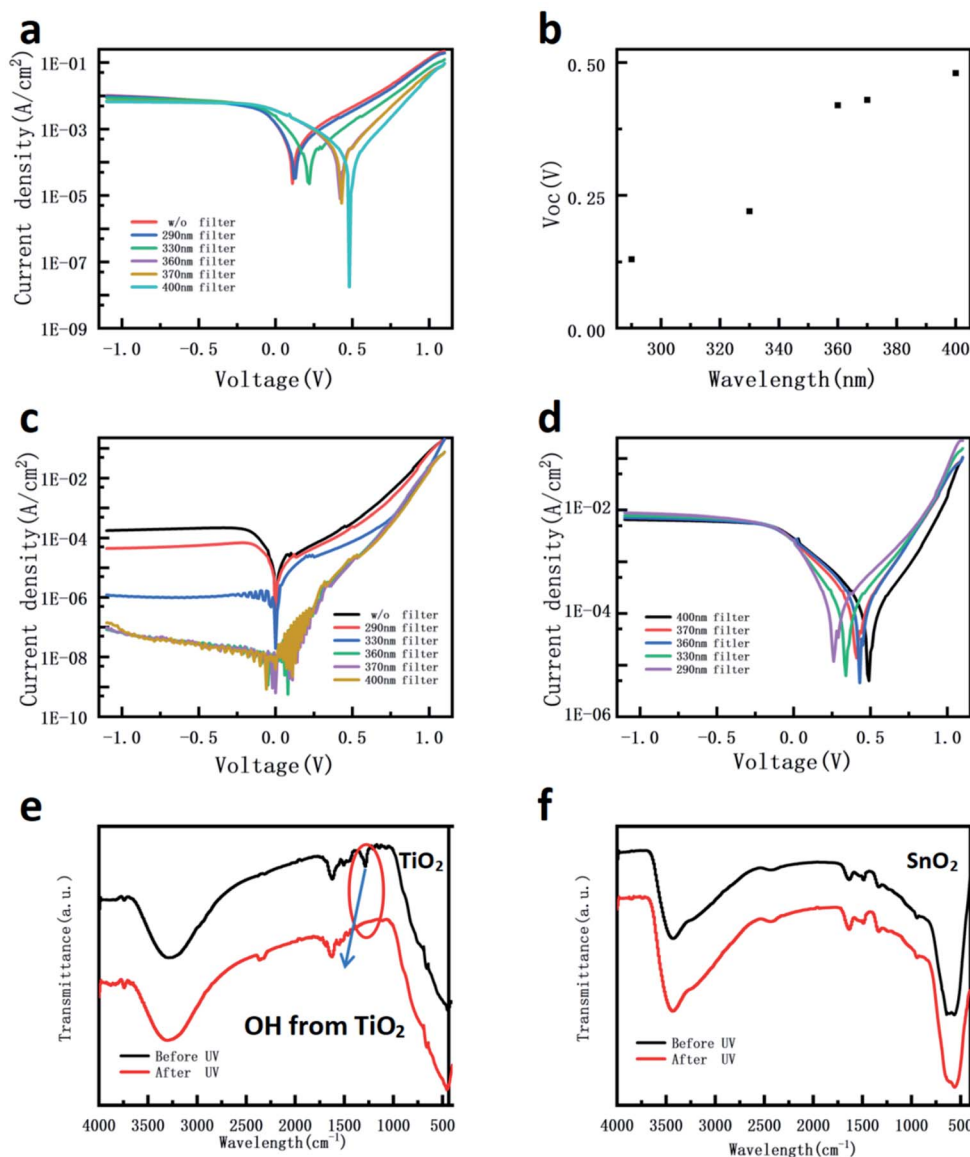


Fig. 4 (a and b) The  $J$ - $V$  curves and specific  $V_{oc}$  value of the  $\text{TiO}_2$  device under different light conditions; (c) the dark currents of the devices after the different light conditions shown in (a); (d) light currents under illumination with a 400 nm filter of the devices after the different exposure conditions shown in (a); the Fourier transform infrared spectroscopy of (e)  $\text{TiO}_2$  and (f)  $\text{SnO}_2$  nanoparticles: red line (black line) is the FTIR spectroscopy of the film after (before) UV illumination for 5 min.

It is well known that the UV absorption of the  $\text{TiO}_2$  nanoparticle is highly effective.<sup>35</sup> Through spectroelectrochemical methods, Donald Fitzmaurice's group has concluded that the surface states of anatase  $\text{TiO}_2$  are located at about 0.5 eV below the conduction band.<sup>36,37</sup> Other works have also shown the existence of a trap state below the conduction band.<sup>38-40</sup> Many groups have reported that the UV exposure can positively shift the conduction band of the  $\text{TiO}_2$  NP film.<sup>41-44</sup> It has been demonstrated that the filling of surface states or UV exposure makes the  $\text{TiO}_2$  NP film more conductive,<sup>39,45-47</sup> and reduces the barrier height between ITO and the  $\text{TiO}_2$  nanoparticle.<sup>48-50</sup> Therefore, the variation of the output characteristics and dark current of the  $\text{TiO}_2$  device after the illumination is most likely

caused by the UV exposure/filling surface state-induced change of the energy band and conductivity of the  $\text{TiO}_2$  NP film.

Fig. 4e and f show the FTIR spectroscopy of the  $\text{TiO}_2$  and  $\text{SnO}_2$  NP films before and after 5 min UV exposure, respectively. The vibration absorption peaks between 3000 and 3800  $\text{cm}^{-1}$  and 1200-1800  $\text{cm}^{-1}$  correspond to the hydroxyl in the  $\text{H}_2\text{O}$  molecule and hydroxyl on the  $\text{TiO}_2$  surface, respectively.<sup>51</sup> The results show that the peak attributed to the hydroxyl on the  $\text{TiO}_2$  surface disappears in the  $\text{TiO}_2$  NP film after UV illumination for 5 min. However, there is no obvious peak change in the  $\text{SnO}_2$  NP film, indicating that the UV light does not make an obvious change in the  $\text{SnO}_2$  NP film. The Ti-OH bond plays an important role in the surface states of the  $\text{TiO}_2$  nanoparticle.<sup>52</sup> The

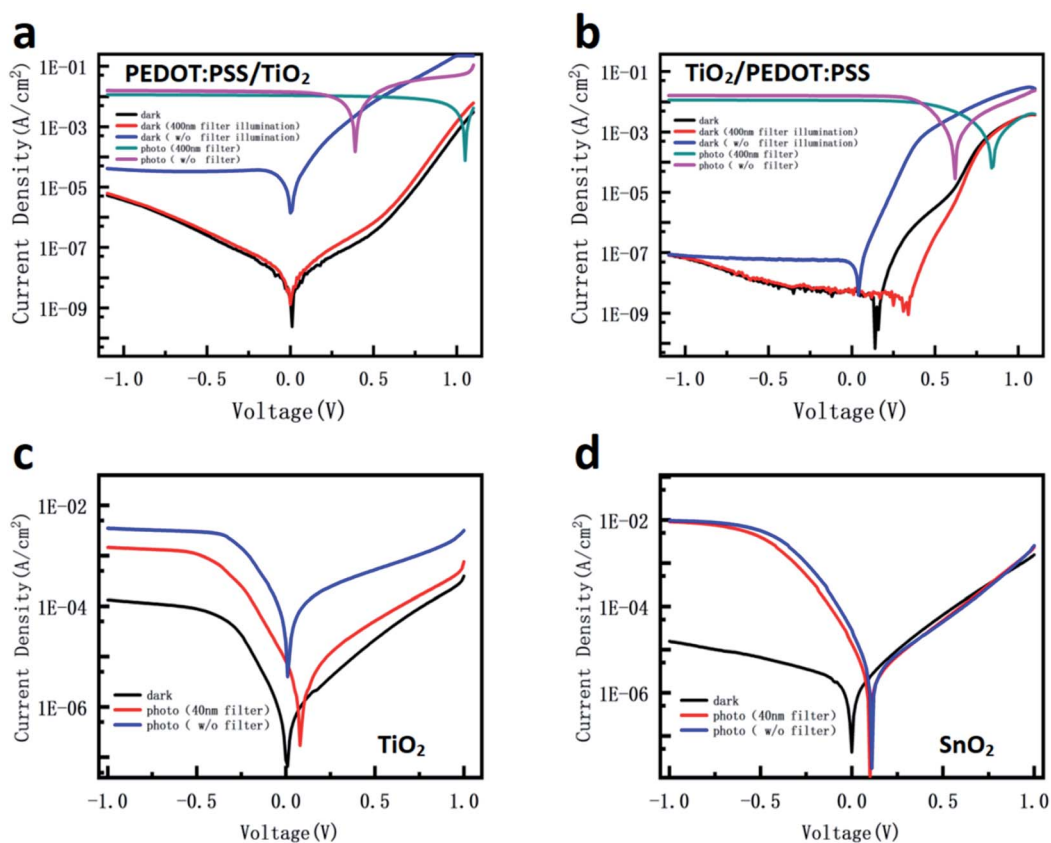


Fig. 5 (a and b)  $J$ - $V$  curves of the ITO/PEDOT:PSS/TiO<sub>2</sub>/perovskite device and the ITO/TiO<sub>2</sub>/PEDOT:PSS/perovskite device. (c and d)  $J$ - $V$  curves for both Al electrode devices: TiO<sub>2</sub> device and SnO<sub>2</sub> device.

oxygen molecule absorbed by hydroxyl and oxygen vacancies are considered as defects, which can trap free electrons from the conduction band of the TiO<sub>2</sub> nanoparticle, and the conductivity of the metal oxide film thus decreases.<sup>52–54</sup> The decomposition of the hydroxyl group reflects the reduction of the surface state, which has been reported in previous works.<sup>55,56</sup> In addition, under UV exposure, the oxygen vacancy-induced surface defects ubiquitous in the metal oxide would trap electrons, leading to the band upshift. Therefore, the hydroxyl on the surface states or oxygen vacancies could influence both energy band and conductivity of TiO<sub>2</sub>.<sup>57–59</sup> After UV treatment, the accumulation of electrons on the surface states or the reduction of defect states enhance the conductivity of TiO<sub>2</sub> since the release of charge become less localized.<sup>60</sup> In addition, the XPS measurements were conducted to evaluate the stability of the metal oxidation states before and after exposure to UV. The results (not shown here) show that the peak positions of the metal in both TiO<sub>2</sub> and SnO<sub>2</sub> films do not change significantly. This observation is consistent with the previous report that demonstrated that the oxidation state of titanium does not change significantly before and after UV.<sup>61</sup>

In addition, because the interface between TiO<sub>2</sub> and perovskite would influence the dissociation of carriers and the injection of electrons,<sup>62</sup> the interface effect on the dark current of the device needs to be clarified. PEDOT:PSS as a common

hole transportation material was inserted between ITO/TiO<sub>2</sub> or TiO<sub>2</sub>/perovskite to make various devices. Fig. 5a and b show the dark currents and photocurrents of the ITO/PEDOT:PSS/TiO<sub>2</sub>/perovskite device and the ITO/TiO<sub>2</sub>/PEDOT:PSS/perovskite device. The former has an obviously higher dark current compared with the latter, implying that the interface between ITO and TiO<sub>2</sub> is another key to keep the low dark current. After the illumination without UV, both device dark currents remain the same as before. However, after the illumination with UV, the dark current of the ITO/TiO<sub>2</sub>/PEDOT:PSS device slightly increased, while that for the ITO/PEDOT:PSS/TiO<sub>2</sub> device significantly increased in comparison with the dark current before illumination. The smaller dark current increase of the former device compared with the latter one is due to the insertion of PEDOT:PSS between TiO<sub>2</sub> and perovskite. The most plausible reason is that the filling of the TiO<sub>2</sub> trap states in the former is only contributed by the photo-generated charges in TiO<sub>2</sub>. Meanwhile, in the latter, it is contributed not only by photogenerated charges in TiO<sub>2</sub>, but also by the transferred photogenerated charges from perovskite since there is no interfacial layer between TiO<sub>2</sub> and perovskite. Greater filling of the surface defects leads to a bigger change of energy band and conductivity of TiO<sub>2</sub>, resulting in a higher dark current. In addition, the function of the difference between the two electrodes was investigated. When the ITO electrode is changed to



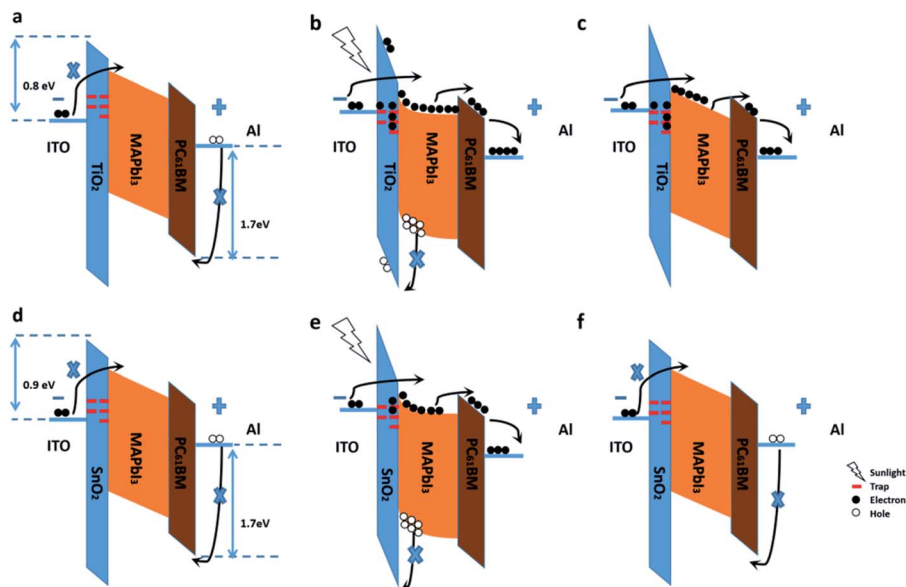


Fig. 6 The working principle schematics: the energy diagram of the  $\text{TiO}_2$  device: (a) in the dark, (b) under illumination, (c) in the dark after illumination; and the energy diagram of the  $\text{SnO}_2$ : (d) in the dark, (e) under illumination, (f) in the dark after illumination.

a transparent Al film, both  $\text{SnO}_2$  and  $\text{TiO}_2$  devices show a large increase of the dark current, as shown in Fig. 5c and d. The results further confirm that the Schottky barrier between the electrode and oxide metal contributed to the inhibition of charge injection.

Finally, the mechanism is proposed as shown in Fig. 6. Under dark condition, the hole injection from the Al electrode and the electron injection from the ITO electrode are both significantly suppressed due to the high charge barriers (1.7 eV between Al and the valence band of  $\text{PC}_{61}\text{BM}$ , 0.8/0.9 eV between ITO and the conduction band of  $\text{TiO}_2/\text{SnO}_2$ ). Besides, the surface states of the  $\text{TiO}_2/\text{SnO}_2$  NPs make the device remain at low conductivity, resulting in a low device dark current. Under illumination, free charge carriers are generated in the perovskite layer, the photo-generated holes are accumulated at the interface between the metal oxide and perovskite due to the interfacial barrier. The accumulated holes at the interface result in band bending, and the conductivity of the metal oxide NP film increases when the trap states of the nanoparticles are occupied by the photo-generated charge carriers. Therefore, a large number of electrons can be injected into the device from the ITO electrode, leading to a significantly larger photocurrent. In addition, unlike  $\text{SnO}_2$  NPs, the decrease in the hydroxyl release-induced trap state density and the trapped photogenerated carriers in the  $\text{TiO}_2$  NPs result in a long-term energy band shift and conductivity increase after UV illumination. Therefore, the dark current of the  $\text{TiO}_2$  device significantly increases, while the  $\text{SnO}_2$  is still as low as the one before illumination.

## Conclusions

In this work, we demonstrate a high-performance PPD using metal oxide nanoparticles as the hole-blocking layer on top of

ITO and  $\text{PC}_{61}\text{BM}$  as the hole-blocking layer under the Al electrode. The resulting significantly low dark current leads to a high on/off ratio and good performances for both  $\text{TiO}_2$  and  $\text{SnO}_2$  devices. After UV illumination, the dark current of the  $\text{TiO}_2$  device dramatically increases, while the dark current of the  $\text{SnO}_2$  device remains the same as before. Therefore, the  $\text{TiO}_2$  device is suitable for vis-NIR photodetector applications, while the  $\text{SnO}_2$  device is good for UV-vis-NIR photodetector applications. This novel device structure paves a new way for high-performance perovskite photodetector fabrications.

## Experimental section

### Materials

$\text{TiCl}_3$  (15–20% basis in HCl) and absolute ethanol were purchased from Aladdin. The  $\text{PbI}_2$  (99.99%), MAI, HPLC water, *N*-dimethylformamide (DMF, 99.8%), and  $\text{SnO}_2$  colloid precursor (15% in  $\text{H}_2\text{O}$  colloidal dispersion) were purchased from Alfa Aesar.  $\text{PC}_{61}\text{BM}$  (99.5%) was purchased from Nano-C company.

### Device fabrication

The ITO-coated glass substrates were pre-cleaned with deionized water and then ethanol in an ultrasonic bath for 30 minutes each time. Titanium trichloride was used as the titanium source to synthesize anatase  $\text{TiO}_2$  nanoparticles. The mixture of 60 mL ethanol, 2 mL  $\text{TiCl}_3$  and 1 mL HCl (6 M) was sealed to control the speed of the hydrolysis reaction, and were heated in the oven at 75 °C for around 8–10 h. The deposition was washed by anhydrous ethanol three times and the final products were dispersed into water for device fabrication.  $\text{PbI}_2$  and MAI were dissolved in DMF as a common mole ratio 1.05 : 1 (the excess  $\text{PbI}_2$  was used for defect passivation and enhancing



crystallization),<sup>63,64</sup> with a concentration of 1.25 M and then stirred at 70 °C overnight. 20 mg PC<sub>61</sub>BM was dissolved in 1 mL chlorobenzene for over 8 h. The ultrasonic treatment of the solution was conducted before the device was fabricated.

The clean indium tin oxide (ITO)-coated glass substrates were treated in UV-ozone for 15 min before deposition of the SnO<sub>2</sub> or TiO<sub>2</sub> layer. The TiO<sub>2</sub> solution (20 mg mL<sup>-1</sup>) was spin-coated at 2000 rpm for 50 s and the formed film was annealed at 130 °C for 10 minutes. The SnO<sub>2</sub> solution (around 200 mg mL<sup>-1</sup>) was spin-coated at 7000 rpm for 40 s, and no further treatment was needed when the film was formed. The perovskite solution was deposited onto the prepared TiO<sub>2</sub> or SnO<sub>2</sub> film with a spin-coating speed of 2500 rpm for 10 s. During the spin coating process, 150 mL anti-solvent (chlorobenzene) was poured onto the spinning substrate to accelerate crystallization. The spin coating speed increased to 5000 rpm for 30 s. The formed perovskite films were heated at 100 °C for 10 minutes, resulting in the completion of crystallization. Furthermore, PC<sub>61</sub>BM (20 mg mL<sup>-1</sup>, dissolved in chlorobenzene) was spin-coated on the active layer at 1000 rpm for 30 s. Finally, a 100 nm thick aluminum film was deposited as the top electrode by thermal evaporation. In addition, a PEDOT:PSS layer can be fabricated by spin-coating a 60 μL precursor on the ITO substrate at 3000 rpm for 40 s and annealed at 140 °C for 10 min. Each device has an effective area of 4.5 mm<sup>2</sup> (3 × 1.5 mm).

### Characterization

The current density *versus* voltage (*J*-*V*) characteristic was measured with a Keithley 2450 instrument in a glovebox filled with N<sub>2</sub>. An Air Mass 1.5 Global (AM 1.5 G) solar simulator was used as the sunlight source with an irradiation intensity of 100 mW cm<sup>-2</sup>. The external quantum efficiency (EQE) was measured by a Zolix Solar Cell Scan 100 in the air. Different wavelength filters were used in front of the light outlet to cut off the short wavelength components to remove unwanted radiation. The Fourier transform infrared (FTIR) spectrum was measured with a Thermo Scientific Nicolet 6700 and the SnO<sub>2</sub> or TiO<sub>2</sub> solution was spin-coated onto the KBr substrate. For the transit photocurrent measurement, a square pulse optical excitation of 1 ms using a 530 nm LED driven by a WF 1946B multifunction synthesizer (NF Corporation) was used. In addition, an IT6133B DC power supply was used to provide different voltages and a NOVA II optical power meter was used to measure the light intensity.

### Conflicts of interest

The authors declare no conflict of interest.

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