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# A confinement of N-heterocyclic molecules in a metal–organic framework for enhancing significant proton conductivity†

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A series of N-heterocyclic/VNU-23 materials have been prepared via the impregnation procedure of N-heterocyclic molecules into VNU-23. Their structural characterizations, PXRD, FT-IR, Raman, TGA, <sup>1</sup>H-NMR, SEM-EDX, and EA, confirmed that N-heterocyclic molecules presented within the pores of parent VNU-23, leading to a remarkable enhancement in proton conductivity. Accordingly, the composite with the highest loading of imidazole, Im<sub>13.5</sub>/VNU-23, displays a maximum proton conductivity value of  $1.58 \times 10^{-2} \text{ S cm}^{-1}$  (85% RH and 70 °C), which is ~4476-fold higher than H<sup>+</sup>/VNU-23 under the same conditions. Remarkably, the proton conductivity of Im<sub>13.5</sub>/VNU-23 exceeds the values at 85% RH for several of the reported high-performing MOF materials. Furthermore, Im<sub>13.5</sub>/VNU-23 can retain a stable proton conductivity for more than 96 h, as evidenced by FT-IR and PXRD analyses. These results prove that this hybrid material possesses potential applications as a commercial proton exchange membrane fuel cell.

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## 1. Introduction

Electrolyte materials have attracted much attention through transforming chemical energy into electric energy by chemical reactions. Currently, proton exchange membrane fuel cells (PEMFCs) are recognized as a new electrolyte material capable of clean and high-energy conversion.<sup>1–4</sup> Nafion, a perfluorosulfonic acid membrane, is extensively employed as a commercial PEMFC due to its high proton conductivity ( $\sigma = 0.1 \text{ S cm}^{-1}$  at 98% relative humidity (RH) at 80 °C).<sup>5</sup> Nevertheless, the synthesis process for Nafion is complex, leading to limitations for scale-up.<sup>6–8</sup> Additionally, the high RH value (98%) requires a high cost to retain during long-term operation and drives flooding at the cathode, resulting in a performance loss of PEMs with only a minor temperature fluctuation.<sup>1</sup> To cope with these challenges, it is necessary to find novel alternative proton conductive materials by incorporating factors as (1) high proton conductive performance at low relative humidity (RH <98%), (2) extended operating time, (3) high chemical and thermal stability, and (4) effortless large-scale.<sup>9–11</sup>

Metal–organic frameworks (MOFs) are porous coordination materials generated from metal clusters and organic bridge linkers with many applications in adsorption and environmental treatment,<sup>12–17</sup> catalysis,<sup>18–20</sup> sensors,<sup>21</sup> and drug delivery.<sup>22,23</sup> Recently, MOFs have been proven to be promising candidates for proton transport with high proton conductivity.<sup>24–26</sup> Besides, the future PEMFCs are desired to act under high voltage and long-term conditions. To fulfill these lofty targets, there is a requirement for creating a new material class with high chemical and mechanical stability. Zirconium-based MOFs are an efficient response material because of contain robust metal–carboxylate bonds and rigid frameworks, thus opening the unique chances for using Zr-based MOFs as the expected proton conductive membrane. As reported, Zr–MOFs exhibited the proton conductivity reached the value of  $10^{-3}$  to  $10^{-2} \text{ S cm}^{-1}$ .<sup>27–30</sup>

The fact revealed that the high-level proton movement relating to the enormous number of protons significantly enhanced the proton conductivity in MOFs. This is addressed through modifying the void space of the frameworks and organic linkers by functional groups (*e.g.*, –COOH, –OH, and –SO<sub>3</sub>H)<sup>29–33</sup> or introducing guest molecules such as ammonia,<sup>34,35</sup> water,<sup>36</sup> and N-heterocyclic molecules (*e.g.*, triazole,<sup>37</sup> histamine,<sup>30,38</sup> and imidazole<sup>29,39,40</sup>). The imidazole derivatives are chosen as the efficient proton carrier agents due to their high polarity, which can favourably uptake water molecules from the external medium, leading to the optimal proton conduction pathway formation.<sup>41–43</sup> However, proton conductivity of imidazole derivatives@MOF composites is

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reduced throughout extended behaviour intervals due to the release of N-heterocyclic molecules from the structure, even resulting in cathode contamination.<sup>26,44</sup> This is attributed to the lack of robust, appropriate interactions between N-heterocyclic molecules and the framework. Hence, it is urgent to seek substitute solutions for minimizing the loss of mentioned guest molecules from the MOF composites.

To retain durable and extended proton conductivity, in our previous work,<sup>29</sup> we suggested an anchoring strategy of imidazole molecules onto VNU-17, constructed from  $Zr_6O_8(H_2O)_8(COO)_8$  cluster and 4-sulfonaphthalene-2,6-dicarboxylic acid linker, containing Brønsted acidic sites of sulfonic groups ( $-SO_3H$ ). VNU-17 is like the harbors to trap the imidazole molecules *via* acid–base reaction of  $SO_3H$  groups and coordinated water molecules with imidazole molecules. As a result, the proton conductivity of  $HIm_{11} \subset VNU-17$  reached  $5.93 \times 10^{-3} \text{ S cm}^{-1}$  at 70 °C and 85% RH without any remarkable change of performance for at least 40 h. Inspired by this, we expect that if the amount of  $SO_3H$  is enhanced within Zr–MOFs, it will increase the anchoring possibility of imidazole derivative molecules into the structure through robust acid–base interactions, assisting an increase of the proton conductivity and prolongation of operating time in the composite membranes. This is a new approach to coping with the stated obstacles about releasing imidazole derivatives of the reported composites during the proton conduction. Additionally, the role of each N-heterocyclic molecule in the proton transfer process is also considered for selecting the most effective proton carrier agent.

Owing to this viable approach, we performed a research strategy, including (i) syntheses of Zr–MOF with high thermal and chemical strength by mixing  $Zr^{4+}$  salt and 4,8-disulfonaphthalene-2,6-dicarboxylic acid linker ( $H_4SNDC$ ) containing two of  $SO_3H$  groups per a linker unit, termed VNU-23,<sup>30</sup> (ii) confinement of a series of N-heterocyclic molecules such as imidazole (Im), 2-methylimidazole (mIm), and benzimidazole (bIm) onto VNU-23 to obtain the hybrid materials as  $Im_x \subset VNU-23$ ,  $mIm_y \subset VNU-23$ , and  $bIm_z \subset VNU-23$ , respectively, (iii) measurement of proton conductivity for these hybrid materials under low RH (<98%) and testing their performing intervals. As expected,  $bIm_{3.7} \subset VNU-23$  and  $mIm_{9.2} \subset VNU-23$  possess the maximum proton conductivity reached the value of  $2.13 \times 10^{-4} \text{ S cm}^{-1}$  and  $4.21 \times 10^{-3} \text{ S cm}^{-1}$  under 85% RH at 70 °C, respectively. Noteworthy, the proton conductivity of  $Im_{13.5} \subset VNU-23$  is  $1.58 \times 10^{-2} \text{ S cm}^{-1}$  (85% RH and 70 °C), which is ~1445-fold higher than that achieves for parent  $H^+ \subset VNU-23$  ( $\sigma = 1.10 \times 10^{-5} \text{ S cm}^{-1}$  under 90% RH at 70 °C). Particularly, the high proton conductivity value of  $Im_{13.5} \subset VNU-23$  can retain more than 96 h, which demonstrates the future applications of this composite material as commercial PEMFCs.

## 2. Experimental

### 2.1 Chemicals and general procedures

Zirconium oxychloride octahydrate ( $ZrOCl_2 \cdot 8H_2O$ , 98%), *N,N*-dimethylformamide (DMF, 99%), formic acid (HCOOH, 95%), imidazole (Im, 99%), 2-methylimidazole (mIm, 99%), benzimidazole (bIm, 99%), oleum ( $SO_3$  in  $H_2SO_4$ , 50%), hydrochloric

acid (HCl, 37%), and anhydrous methanol (MeOH, 99%) were purchased from local vendors and employed without further purification. 4,8-Disulfonaphthalene-2,6-dicarboxylic acid ( $H_4SNDC$ ) was prepared according to the previously reported synthetic method.<sup>30</sup>

Fourier transform infrared spectroscopy (FT-IR) measurements were carried out on a Jasco spectrometer with the Attenuated Total Reflectance (ATR) sampling method. Raman spectroscopy analysis was collected on a Horiba XploRA ONE 532 nm. Thermal gravimetric analyses (TGA) were measured on a LabSys Evo thermal analysis system under dry air flow and in the temperature range of 25–800 °C. Powder X-ray data were recorded on a Bruker D8 Advance using Ni filtered  $Cu K\alpha$  ( $\lambda = 1.54718 \text{ \AA}$ ). Water adsorption and desorption curves were analyzed on a Belsorp-aqua3 instrument at ambient temperature.  $^1H$ -NMR spectra were obtained on a Bruker Advance NEO-600 MHz NMR spectrometer. Humidity was controlled by an Espec humidity chamber (SH-222). Scanning electron microscopy (SEM) image was performed on a Hitachi FESEM S-4800 microscope with accelerating voltage of 10 kV. Energy-dispersive X-ray (EDX) analysis was measured on a Horiba EDX H-7593 instrument. Elemental analyses of C, H, N, and S were performed using a LECO CHNS-932 analyzer. Low pressure  $N_2$  adsorption analyses were analyzed on a Micromeritics 3Flex surface characterization analyzer.

The materials (120 mg) were ground and pelletized for the alternating current (ac) impedance spectroscopy analyses, which were conducted on a Gamry potentiostat Interface 1000 with the two-electrode method, and the frequencies ranged from  $10^6$  to 10 Hz with the applied voltage altered from 1 to 10 mV. The samples were pressed at 4.2 MPa with 13 mm diameter. The thickness of the pellets was analyzed by a Nikon SMZ1000 microscope, varying from 0.60 to 0.62 mm. The impedance spectra were collected under different RHs and temperatures, in which the stability of relative humidity was maintained using the saturated salt solutions placed inside the chamber. The pellets were balanced for at least 24 h at different temperatures and RH values to retain the persistent RH before measurements for all the ac impedance spectra. The proton conductivity was calculated using the equation  $\sigma = L/SR$ , where  $\sigma$ ,  $L$ ,  $S$  are the conductivity ( $\text{S cm}^{-1}$ ), the thickness of the samples (cm), the surface area of the pellets ( $\text{cm}^2$ ), respectively and  $R$  is the resistance of membrane ( $R_2$ ) counted by fitting out the Nyquist plots using the equivalent circuit diagram (Fig. S8†) through Gamry Echem Analyst program.

### 2.2 Synthesis of $H^+ \subset VNU-23$

According to the previously reported work,<sup>30</sup>  $H^+ \subset VNU-23$  was prepared by the solvothermal method. A mixture of  $H_4SNDC$  (0.825 g, 2.183 mmol), and  $ZrOCl_2 \cdot 8H_2O$  (0.650 g, 2.025 mmol) was introduced into a 200 mL glass bottle including HCOOH and DMF (25 mL and 100 mL, respectively). The mixture was then sonicated for 10 min and heated at 120 °C for 48 h. Subsequently, the mixture was cooled to room temperature, yielding to the powder, namely pristine VNU-23. This product was washed by DMF ( $2 \times 50 \text{ mL}$ ) for 72 h. The sample was



exchanged with a solution ( $10 \times 10$  mL), containing  $\text{H}_2\text{SO}_4$  (0.3 M) with MeOH/ $\text{H}_2\text{O}$  solvent ( $v/v = 4/1$ ) for 48 h. Next, the product was centrifuged, and washed by an excess amount of MeOH/ $\text{H}_2\text{O}$  ( $v/v = 4/1$ ) to obtain a solution with pH = 5. Finally, the material was exchanged with MeOH ( $6 \times 20$  mL) for 48 h, dried, and activated under vacuum at  $80^\circ\text{C}$  for 24 h to yield  $\text{H}^+\text{cVNU-23}$ .

### 2.3 Synthesis of $\text{Im}_{13.5}\text{cVNU-23}$

The activated  $\text{H}^+\text{cVNU-23}$  (100 mg) was soaked in 10 mL of MeOH solution of Im (5 M). The mixture was stirred at  $40^\circ\text{C}$  for 48 h. Continuously, 10 mL of fresh MeOH solution of Im (5 M) was introduced the mixture to assure the saturated loading of Im onto  $\text{H}^+\text{cVNU-23}$  and stirred for 24 h at  $40^\circ\text{C}$ . Then, the sample was centrifuged, and washed with a copious amount of MeOH ( $3 \times 2$  mL) to remove Im molecules on the surface and within intercrystalline regions. The material was activated at room temperature under vacuum for 24 h to acquire  $\text{Im}_{13.5}\text{cVNU-23}$ . FT-IR ( $\text{cm}^{-1}$ , ATR): 3130 (w), 1607 (m), 1571 (s), 1414 (s), 1360 (s), 1321 (m), 1184 (s), 1065 (m), 1035 (s), 927 (w), 812 (m), 763 (s), 658 (s), 612 (s). Anal. calcd (%) for  $\text{C}_{88.5}\text{H}_{118.6}\text{O}_{68.3}\text{N}_{27}\text{S}_8\text{Zr}_6 = \{\text{Zr}_6\text{O}_8(\text{H}_2\text{O})_8(\text{C}_{12}\text{H}_6\text{O}_{10}\text{S}_2)_4\} \cdot 9.2(\text{C}_4\text{H}_6\text{N}_2) \cdot 7.7\text{H}_2\text{O}$ : C, 30.75; H, 3.43; N, 10.95; S, 7.41. Found: C, 31.19; H, 3.52; N, 11.21; S, 7.02.

### 2.4 Synthesis of $\text{mIm}_{9.2}\text{cVNU-23}$

The activated  $\text{H}^+\text{cVNU-23}$  (100 mg) was immersed in a MeOH solution of mIm (10 mL, 5 M). The mixture was stirred at  $40^\circ\text{C}$

for 48 h. Subsequently, 10 mL of fresh MeOH solution of mIm (5 M) was introduced the mixture to obtain the saturated loading of mIm onto  $\text{H}^+\text{cVNU-23}$  and stirred for 24 h at  $40^\circ\text{C}$ . The solid was centrifuged and washed with an excess amount of MeOH ( $3 \times 2$  mL) to remove mIm molecules on the surface and within intercrystalline regions. The product was activated at room temperature under vacuum for 24 h to yield  $\text{mIm}_{9.2}\text{cVNU-23}$ . FT-IR ( $\text{cm}^{-1}$ , ATR): 3146 (w), 1607 (m), 1567 (s), 1418 (s), 1363 (s), 1303 (w), 1180 (m), 1117 (m), 1034 (m), 920 (w), 812 (m), 766 (s), 657 (s), 614 (s). Anal. calcd (%) for  $\text{C}_{84.8}\text{H}_{110.6}\text{O}_{63.7}\text{N}_{18.4}\text{S}_8\text{Zr}_6 = \{\text{Zr}_6\text{O}_8(\text{H}_2\text{O})_8(\text{C}_{12}\text{H}_6\text{O}_{10}\text{S}_2)_4\} \cdot 9.2(\text{C}_4\text{H}_6\text{N}_2) \cdot 7.7\text{H}_2\text{O}$ : C, 31.73; H, 3.45; N, 8.03; S, 7.98. Found: C, 31.95; H, 3.31; N, 7.62; S, 7.66.

### 2.5 Synthesis of $\text{bIm}_{3.7}\text{cVNU-23}$

This composite was prepared with the same method as these materials above. The activated  $\text{H}^+\text{cVNU-23}$  (100 mg) was immersed in a MeOH solution of bIm (10 mL, 5 M). The mixture was stirred at  $40^\circ\text{C}$  for 48 h. Subsequently, 10 mL of fresh MeOH solution of bIm (5 M) was introduced the mixture to acquire the saturated loading of bIm onto  $\text{H}^+\text{cVNU-23}$  and stirred for 24 h at  $40^\circ\text{C}$ . Similarly, the material was centrifuged, and washed with an excess amount of MeOH ( $3 \times 2$  mL) to ensure the removal of bIm molecules on the surface and within intercrystalline regions. The solid was activated at room temperature under vacuum for 24 h to obtain  $\text{bIm}_{3.7}\text{cVNU-23}$ . FT-IR ( $\text{cm}^{-1}$ , ATR): 3101 (w), 1607 (m), 1569 (s), 1417 (s), 1364 (s), 1298 (w), 1179 (m), 1119 (m), 1036 (s), 924 (w), 812 (m), 767 (s), 658 (s), 611 (s). Anal. calcd (%) for  $\text{C}_{66.9}\text{H}_{87.4}\text{O}_{71.6}\text{N}_{5.4}\text{S}_8\text{Zr}_6 =$

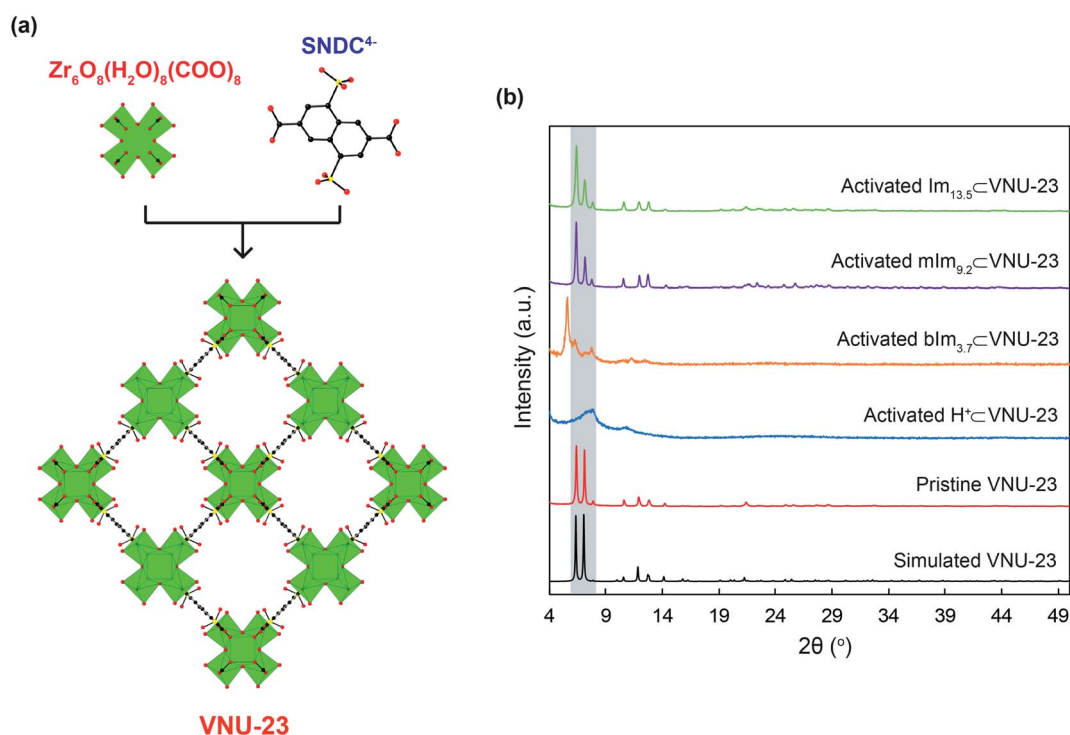


Fig. 1 (a) The structure of VNU-23 backbone is constructed from  $\text{Zr}_6\text{O}_8(\text{H}_2\text{O})_8(\text{COO})_8$  and ditopic  $\text{SNDC}^{4-}$  linkers<sup>30</sup> and (b) PXRD pattern of simulated VNU-23 (black) in comparison with pristine VNU-23 (red), activated  $\text{H}^+\text{cVNU-23}$  (blue),  $\text{bIm}_{3.1}\text{cVNU-23}$  (orange),  $\text{mIm}_{9.2}\text{cVNU-23}$  (purple), and  $\text{Im}_{13.5}\text{cVNU-23}$  (green). Atom colors: Zr, green polyhedra; C, black; O, red; S, yellow. All H atoms are omitted for clarity.



$\{Zr_6O_8(H_2O)_8(C_{12}H_6O_{10}S_2)_4\} \cdot 3.7(C_7H_6N_2) \cdot 15.6H_2O$ : C, 29.25; H, 3.08; N, 3.42; S, 8.44. Found: C, 29.41; H, 2.95; N, 3.22; S, 8.17.

### 3. Results and discussion

#### 3.1 Synthesis and characterization

Based on the previously published study,<sup>30</sup> VNU-23 was generated by blending  $ZrOCl_2 \cdot 8H_2O$  salt and  $H_4SNDC$  linker in DMF solvent with formic acid as a modular for orientating the cluster structure and supporting the crystal nucleation. Subsequently, the mixture was heated at 120 °C for 48 h. In an attempt to understand the confinement mechanism of N-heterocyclic molecules onto VNU-23 for achieving the optimal proton transport pathway, there is a need to recall the structure of VNU-23. VNU-23 crystallizes in the space group of  $I4/m$  and its unit cell parameter is  $a = b = 17.7174 \text{ \AA}$  and  $c = 22.4865 \text{ \AA}$ .<sup>30</sup> Four connected  $SNDC^{4-}$  linkers and 8-connected  $Zr_6O_8(H_2O)_8(COO)_8$  clusters construct the VNU-23 framework. This leads to the generation of a three-dimensional network framework with *bcu* topology (Fig. 1a). Herein, dimethylammonium ions is created during the synthetic process of VNU-23, which incorporate with

$SO_3^-$  ions of linkers to obtain the rigid framework, assisting the conformity of the structure in comparison with the simulated VNU-23 (Fig. 1b). In order to recover the  $SO_3H$  groups, pristine VNU-23 was immersed in  $H_2SO_4$  solution (0.3 M) and washed until the liquor reached  $pH = 5$  to acquire  $H^+ \subset VNU-23$ . The activated  $H^+ \subset VNU-23$  is lost the structural order due to the versatility of  $SO_3H$  after rehabilitation.<sup>45,46</sup> Notwithstanding, the crystallinity is recovered as soaked in water and MeOH (Fig. S1†). With this phenomenon, we assume that  $H^+ \subset VNU-23$  possesses the packed sulfonic groups ( $SO_3H$ ), which can facilitate for doping process of the imidazole derivatives within the pore *via* acid–base reaction (Fig. 2), leading to the formation of the rigid backbone to maintain the structural order. Accordingly, N-heterocyclic  $\subset VNU-23$  materials were synthesized by using the activated  $H^+ \subset VNU-23$  adsorbed the solutions of Im, mIm, and bIm at 40 °C. The PXRD analyses were assigned as the original confirmations for the successful doping of N-heterocyclic molecules in the pores of  $H^+ \subset VNU-23$ . Fig. 1b reveals that the PXRD patterns of activated  $Im_{13.5} \subset VNU-23$  and  $mIm_{9.2} \subset VNU-23$  are in good agreement with simulated VNU-23. This demonstrates the effective encapsulation of Im and

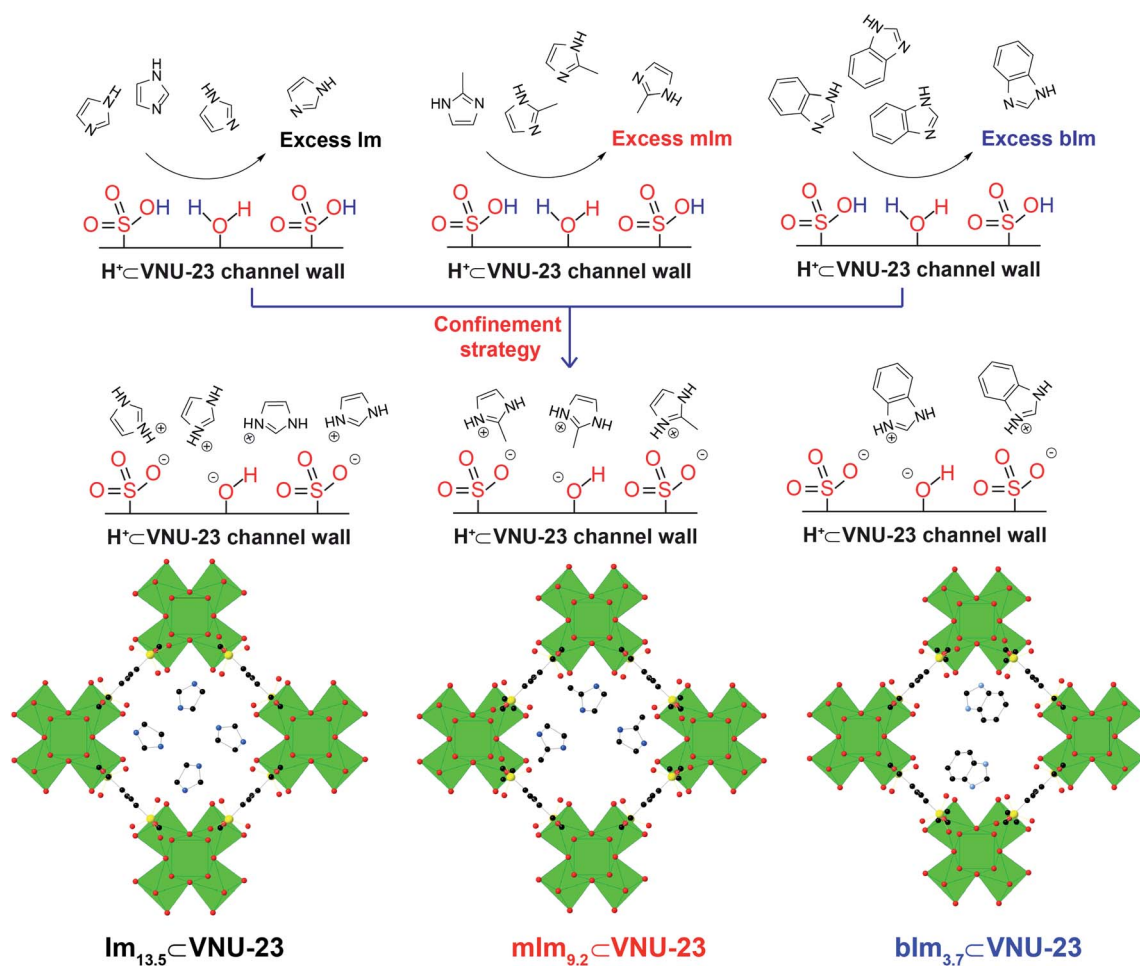


Fig. 2 Confinement strategy of N-heterocyclic molecules on the channel wall of  $H^+ \subset VNU-23$ : plausible mechanism and the assumed position for encapsulating N-heterocyclic molecules on the channel wall of  $H^+ \subset VNU-23$ . Atom colors: Zr, green polyhedra; C, black; O, red; S, yellow. All H atoms are omitted for clarity.



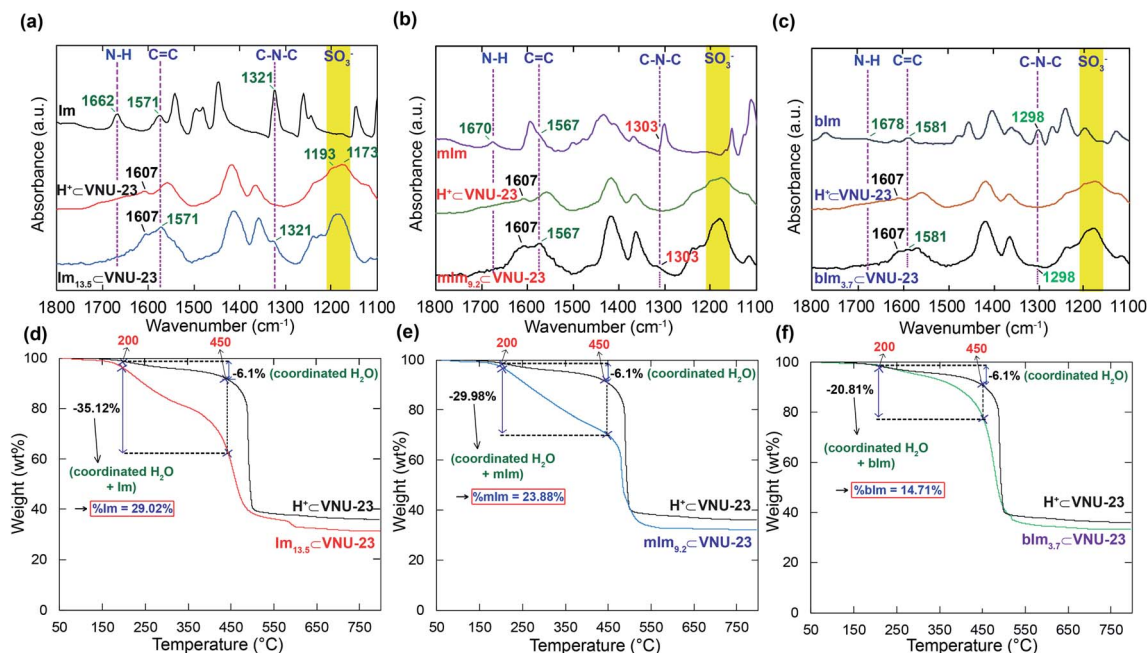


Fig. 3 Fourier transform-infrared spectroscopy analysis of N-heterocyclic VNU-23: (a)  $\text{Im}_{13.5}\text{-VNU-23}$ , (b)  $\text{mIm}_{9.2}\text{-VNU-23}$ , (c)  $\text{bIm}_{3.7}\text{-VNU-23}$  \* VNU-23 in comparison with  $\text{H}^+\text{-VNU-23}$ ; thermal gravimetric analysis of N-heterocyclic VNU-23 materials as compared to  $\text{H}^+\text{-VNU-23}$  (d–f).

$\text{mIm}$  onto  $\text{H}^+\text{-VNU-23}$ . In contrast, there is a deviation in the PXRD pattern of  $\text{bIm}_{3.7}\text{-VNU-23}$  in comparison with the simulated VNU-23 (Fig. 1b). It can be explained that the molecular size of  $\text{bIm}$  is quite bulky, driving to its poor loading into the pores. Noteworthy, the content-dependent and time-dependent amounts of adsorbed  $\text{Im}$ ,  $\text{mIm}$ , and  $\text{bIm}$  were also observed.

The maximum anchoring amounts of N-heterocyclic molecules were obtained at the adsorbing concentration of 5 M and time of 72 h (Fig. S4 and S5<sup>†</sup>). The quantities of  $\text{Im}$ ,  $\text{mIm}$ , and  $\text{bIm}$  molecules anchored into the pores of  $\text{H}^+\text{-VNU-23}$  were estimated by elemental analysis (EA). Subsequently, the Fourier transform infrared (FT-IR) spectra of  $\text{Im}_{13.5}\text{-VNU-23}$ ,  $\text{mIm}_{9.2}\text{-VNU-23}$ , and  $\text{bIm}_{3.7}\text{-VNU-23}$  are exhibited in Fig. 3a–c, distinguished with the IR spectrum of  $\text{H}^+\text{-VNU-23}$  and the corresponding  $\text{Im}$  derivatives with the new signals located at 1298–1321  $\text{cm}^{-1}$  and 1567–1581  $\text{cm}^{-1}$ , which appear in the  $\text{Im}$ ,  $\text{mIm}$ , and  $\text{bIm}$  spectra, but absent in  $\text{H}^+\text{-VNU-23}$  range. These bands derive from C–N–C and C=C vibrations of  $\text{Im}$ ,  $\text{mIm}$ , and  $\text{bIm}$  molecules, respectively. Besides, the absence of the adsorption signals at 1662, 1670, and 1678  $\text{cm}^{-1}$  (confirmed to the N–H bending) in  $\text{Im}_{13.5}\text{-VNU-23}$ ,  $\text{mIm}_{9.2}\text{-VNU-23}$ , and  $\text{bIm}_{3.7}\text{-VNU-23}$  materials assign that all N-heterocyclic molecules are totally located within the pores of  $\text{H}^+\text{-VNU-23}$ . This is also evidenced by Raman spectroscopy analysis (Fig. S3<sup>†</sup>). Hence, the washing process of these composites by MeOH has entirely removed the abundant N-heterocyclic molecules from the surface and the intercrystalline regions. Furthermore, N-heterocyclic VNU-23 materials exhibit the unique performance of weight loss from  $\text{H}^+\text{-VNU-23}$ . As given in Fig. 3d–f,  $\text{H}^+\text{-VNU-23}$  shows a slope of weight loss (25–200 °C), and  $\sim 2$  wt% is attributed to the release of

adsorbed water molecules on the surface, followed by a drop from 200 to 450 °C until the framework decomposition. The weight loss (200–450 °C) is  $\sim 6.1\%$  and accounted for the loss of coordinated water from the clusters of  $[\text{Zr}_6\text{O}_8(\text{H}_2\text{O})_8(\text{C}_{12}\text{H}_6\text{O}_{10}\text{S}_2)_4]$ . However, N-heterocyclic VNU-23 composites display a steeper drop (200–450 °C) assigned to releasing N-heterocyclic and coordinated water molecules. This situation is further confirmed by variable-temperature PXRD analyses (Fig. S2<sup>†</sup>), which reveal that the frameworks of  $\text{Im}_{13.5}\text{-VNU-23}$  are thermally stable  $\leq 450$  °C.<sup>29,30</sup> Accordingly, the weight percentages of N-heterocyclic molecules in the range of 200–450 °C (29.02, 23.88, and 14.71% for  $\text{Im}_{13.5}\text{-VNU-23}$ ,  $\text{mIm}_{9.2}\text{-VNU-23}$ , and  $\text{bIm}_{3.7}\text{-VNU-23}$ , respectively) are determined that there are  $\sim 13.9$   $\text{Im}$ , 8.9  $\text{mIm}$ , and 3.4  $\text{bIm}$  molecules per formula unit of  $[\text{Zr}_6\text{O}_8(\text{H}_2\text{O})_8(\text{C}_{12}\text{H}_6\text{O}_{10}\text{S}_2)_4]$  loaded in the framework of  $\text{H}^+\text{-VNU-23}$ ,<sup>29,30,39</sup>. Particularly,  $\text{Im}_{13.5}\text{-VNU-23}$  sample was digested in acidic condition to perform  $^1\text{H-NMR}$  measurements. From the resulting data, an imidazolium/ $\text{H}_4\text{SNDC}$  ratio of 3.41 is determined (Fig. S5<sup>†</sup>). This corresponded to 13.6  $\text{Im}$  molecules per a formula unit of VNU-23. These values are in good agreement with EA result in the experimental section. All of the stated evidence, achieved from EA, PXRD, FT-IR, Raman, TGA, and  $^1\text{H-NMR}$  measurements perceive that the N-heterocyclic molecules have been successfully immobilized into the framework of VNU-23.

### 3.2 Proton conduction of N-heterocyclic VNU-23

Water adsorption/desorption curve is considered the vital factor for proton transfer, which helps us choose a perfect cut-off point for optimal proton conduction pathway to minimize the effect of condensed water on the surface of the compressed



pellets, resulting in the reduction of condensed water in accurate measurements. Hence, previous to measuring the proton conductivity of the composites, the water adsorption/desorption experiments were conducted at 25 °C. As shown in Fig. 4a–e, the water is condensed at  $P/P_o = 0.93$  for pristine VNU-23 and  $H^+ \subset VNU-23$  materials, observed by the steep increase of water uptake. Therefore, their cut-off point is selected at  $P/P_o = 0.9$  (RH = 90%) for measuring their proton conductivity. Meanwhile, all N-heterocyclic  $\subset VNU-23$  composites possess the adsorption curves with the water condensed at  $P/P_o = 0.87$ , thus the proton conductivities of these composites are performed as high as possible at  $P/P_o = 0.85$  (RH = 85%). Particularly, it is noted that the uptake capacity of  $Im_{13.5} \subset VNU-23$  is slightly higher than  $mIm_{9.2} \subset VNU-23$  and  $bIm_{3.7} \subset VNU-23$  at 85% RH, despite the truth that the pores of  $mIm_{9.2} \subset VNU-23$  and  $bIm_{3.7} \subset VNU-23$  possess a greater loading space of water than that of  $Im_{13.5} \subset VNU-23$ . This phenomenon is similar to the previous works,<sup>29,30</sup> and attributed to the total hydration of  $Im_{13.5} \subset VNU-23$  at the grain boundary until the water condensed on the surface of the compressed pellet at higher RH.

Through the structural characterizations of the N-heterocyclic  $\subset VNU-23$  composites with the packed arrangement of sulfonate groups conjugated to the protonated N-heterocyclic molecules, an impedance analyses were conducted on the pelletized samples to determine their proton conductivities.

The proton conductivity of these composites was measured at different RH values (50–85%) with an immobilized temperature of 70 °C and at various temperatures (30–70 °C) with a fixed RH of 85%. As illustrated in Section S6,<sup>†</sup> the Nyquist plots show a semicircle in the high-frequency range caused by bulk and grain boundary, together with a tail at the low-frequency region dealing with the mobile ions blocked by the electrode–electrode interface.<sup>47</sup> Here, the resistance of the materials was calculated from the  $Z'$ -axis intercept parameter.<sup>38</sup> Accordingly, the proton conductivity of  $Im_{13.5} \subset VNU-23$ ,  $mIm_{9.2} \subset VNU-23$ , and  $bIm_{3.7} \subset VNU-23$  is significantly relative to the RH values, which increases together with rising RH value (Fig. 5). It can be ascribed to the variations in the mobile degree of water molecules in pore channels at different RHs. At higher RH, the water uptake capacity of the materials increases, which supports the transfer process of protons through hydrogen bonding networks, leading to the enhancement in conductivity. In addition, the vital role of confined N-heterocyclic molecules within the structure has been demonstrated *via* the maximum proton conductivity of  $1.58 \times 10^{-2} \text{ S cm}^{-1}$  for  $Im_{13.5} \subset VNU-23$  correlated with  $4.21 \times 10^{-3} \text{ S cm}^{-1}$  for  $mIm_{9.2} \subset VNU-23$ , and  $2.13 \times 10^{-4} \text{ S cm}^{-1}$  for  $bIm_{3.7} \subset VNU-23$  at the same condition (85% RH, 70 °C). It is interesting to note that the higher amount of encapsulated Im derivatives increases of proton conductivity in all RH values.

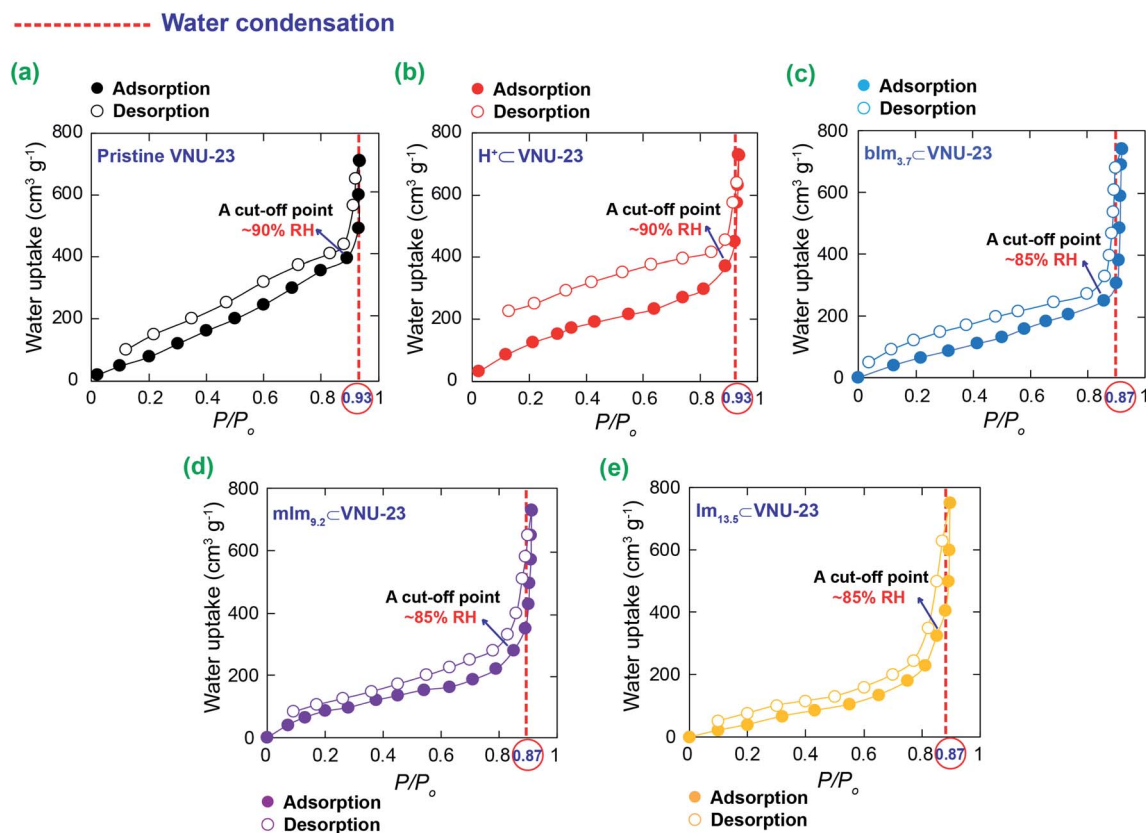


Fig. 4 The water uptake of N-heterocyclic  $\subset VNU-23$  (c, d and e) in comparison with pristine VNU-23 (a) and  $H^+ \subset VNU-23$  (b) at 25 °C as a function of  $P/P_o$ .



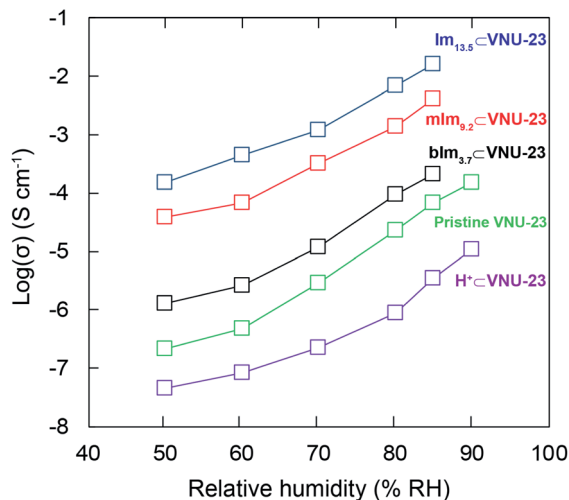


Fig. 5 Dependence of proton conductivity in  $\text{Im}_{13.5}\text{@VNU-23}$  (blue),  $\text{mIm}_{9.2}\text{@VNU-23}$  (red), and  $\text{blm}_{3.7}\text{@VNU-23}$  (black) in comparison with pristine VNU-23 (green) and  $\text{H}^+\text{@VNU-23}$  as a function of relative humidity at 70 °C.

Particularly, the proton conductivity of  $\text{Im}_{13.5}\text{@VNU-23}$  is  $\sim 234$  times higher than pristine VNU-23 with 8 dimethylammonium ions per a VNU-23 unit<sup>30</sup> ( $6.75 \times 10^{-5} \text{ S cm}^{-1}$  under 85% RH and 70 °C) and  $\sim 4476$  times higher than  $\text{H}^+\text{@VNU-23}$  ( $3.53 \times 10^{-6} \text{ S cm}^{-1}$  under 85% RH and 70 °C). The fact shows that the sulfonic acid group, a strong acid, can devote  $\text{H}^+$  ions to the conduction pathways more efficiently than dimethylammonium and imidazolium derivative ions, where  $\text{H}^+$  is attached to the basic nitrogen sites of N-heterocyclic

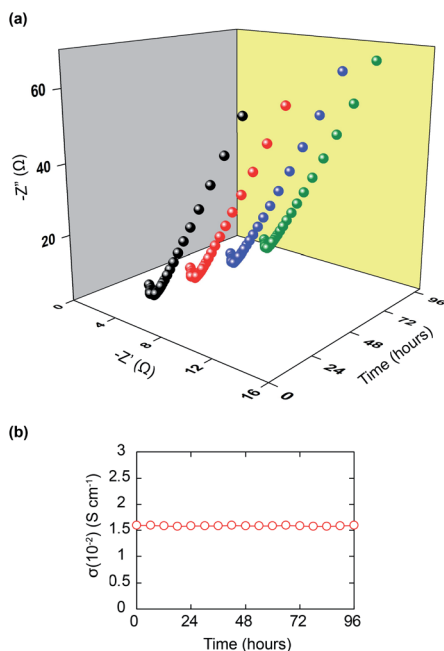


Fig. 6 Time-dependent Nyquist plots of  $\text{Im}_{13.5}\text{@VNU-23}$  (a); time-dependent proton conductivity of  $\text{Im}_{13.5}\text{@VNU-23}$  conducted under 85% RH at 70 °C (b).

Table 1 The proton conductivity values for N-heterocyclic@VNU-23 in comparison with high-performing proton conducting materials

Material	$\sigma$ ( $\text{S cm}^{-1}$ )	Conditions	Ref.
$\text{Im}_{13.5}\text{@VNU-23}$	$1.58 \times 10^{-2}$	70 °C, 85% RH	This work
	$5.82 \times 10^{-3}$	25 °C, 85% RH	
$\text{mIm}_{9.2}\text{@VNU-23}$	$4.21 \times 10^{-3}$	70 °C, 85% RH	
$\text{blm}_{3.7}\text{@VNU-23}$	$2.13 \times 10^{-4}$	70 °C, 85% RH	
$\text{Im/VNU-23}$	$3.04 \times 10^{-6}$	70 °C, 85% RH	
$\text{Im@MOF-808}$	$3.45 \times 10^{-2}$	65 °C, 99% RH	39
	$1.05 \times 10^{-3}$	25 °C, 85% RH	
$\text{Im-Fe-MOF}$	$1.21 \times 10^{-2}$	60 °C, 98% RH	48
	$5.56 \times 10^{-6}$	25 °C, 85% RH	
$\text{HIm}_{11}\text{@VNU-17}$	$5.93 \times 10^{-3}$	70 °C, 85% RH	29
$\text{PCMOF10}$	$7.50 \times 10^{-3}$	70 °C, 85% RH	50
$\text{UiO-66-(CO}_2\text{H)}_2$	$1.25 \times 10^{-4}$	30 °C, 85% RH	45
$\text{UiO-66}$	$3.16 \times 10^{-8}$	30 °C, 85% RH	
$\text{HIm@UiO-67}$	$1.44 \times 10^{-3}$	120 °C	51

molecules, probably resulting in higher conductivity. However, the proton conductivity of pristine VNU-23 and N-heterocyclic@VNU-23 samples is higher than parent  $\text{H}^+\text{@VNU-23}$ , which is attributed to the full hydration of their grain boundary and pore channels addressed by the high polarity of imidazolium ions. This situation is also found in the reported studies.<sup>29,30,39,48</sup>

To further confirm the importance of Im molecules inside VNU-23 for enhancing the proton conductivity, we performed a series of additional experiments. In detail,  $\text{H}^+\text{@VNU-23}$  was mixed with pure Im at the same equivalent as the experimental section to obtain the new composite, termed  $\text{Im/VNU-23}$ . Section S7† reveals that  $\text{Im/VNU-23}$  samples display a poor proton conductivity reached  $\sim 3.04 \times 10^{-6} \text{ S cm}^{-1}$  at 70 °C and 85% RH, which is  $\sim 5197$  times lower than  $\text{Im}_{13.5}\text{@VNU-23}$ . This indicates that if Im molecules locate independently with the framework of VNU-23, they will not join and support favorably for the proton transport process of the material.

Moreover, it is seen that the proton conductivity of  $\text{Im}_{13.5}\text{@VNU-23}$  at 70 °C and 85% RH betters the values at 85% RH for several reported MOF materials (Table 1). Even though  $\text{Im@MOF-808}$  (ref. 39) exhibits the proton conductivity of  $3.45 \times 10^{-2} \text{ S cm}^{-1}$  at 65 °C under.

99% RH, however, this RH value demands a high cost for retaining, and it quickly drives the flood at the cathode only by a minor temperature fluctuation, causing the decrease of proton conductivity. Additionally, investigating practical applications, the time-dependent conductivities of  $\text{Im}_{13.5}\text{@VNU-23}$  have been performed under 85% RH at 70 °C. Fig. 6 shows that the proton conductivity of  $\text{Im}_{13.5}\text{@VNU-23}$  is maintained for at least 96 h, without any remarkable reduction in performance. This indicates the desirable durability property of proton conductivity of  $\text{Im}_{13.5}\text{@VNU-23}$  for prospective applications in PEMFCs. In order to inspect whether the structure of  $\text{Im}_{13.5}\text{@VNU-23}$  changed after measuring long-term ac impedance analyses, FT-IR and PXRD analyses were performed to characterize its stability. As given in Section S8,† the FT-IR spectra and PXRD patterns of  $\text{Im}_{13.5}\text{@VNU-23}$  are consistently



retained before and after ac impedance analyses (96 h), which demonstrate the high structural stability of this composite.

### 3.3 Plausible proton transport mechanism of N-heterocyclic/VNU-23

To understand the proton transfer proton mechanism in N-heterocyclic/VNU-23 materials, the proton conductivities of these composites with the temperature range from 30 to 70 °C under 85% RH are exhibited in Section S6.† It is realized that the proton conductivities of the materials show the trend of increasing with the increasing temperature. This can be explained by the higher temperatures providing energy for the dissociation process of the protons.<sup>40</sup> Besides, the activation energy ( $E_a$ ) values under 85% RH of these composites were determined. The temperature dependence of proton conductivity could be calculated from eqn (1).<sup>49</sup>

$$\ln(\sigma T) = \ln(\sigma_0) - E_a/k_B T \quad (1)$$

Where  $\sigma$  and  $\sigma_0$  are the proton conductivity and the pre-exponential factor,  $k_B$  and  $T$  are the Boltzmann constant and temperature. The activation energy is calculated from the slope of  $1000T^{-1}$  vs.  $\ln(\sigma T)$ .

In general, there are two mechanism types for the proton conduction in PEMFCs according to the activation energy, which reaches 0.1–0.4 eV for the Grotthuss mechanism and 0.4–0.9 eV for the vehicle mechanism.<sup>26</sup> Consequently, the Arrhenius plots of the hybrid materials are shown in Fig. 7. The  $E_a$  values of  $\text{Im}_{13.5}/\text{VNU-23}$  and  $\text{mIm}_{9.2}/\text{VNU-23}$  are 0.208 eV and 0.363 eV, which indicate the proton conduction occurred *via* a Grotthuss mechanism.

This can be ascribed that there are enormous hydrogen-bonding systems formed by external water molecules, coordinated water molecules, imidazolium, and 2-methylimidazolium ions, and sulfonate groups, which are the optimal proton

transfer pathways for facilitating the jump of protons. Meanwhile, the Arrhenius fitting of  $\text{bIm}_{3.7}/\text{VNU-23}$  results from the  $E_a$  of 0.427 eV demonstrates a vehicle mechanism. This may be because  $\text{bIm}$  molecules are loaded into the channel of VNU-23 with a small amount, resulting in the proton conduction through moving directionally into the channel.

## 4. Conclusion

In summary, the hybrid materials were successfully synthesized through a confinement approach of N-heterocyclic molecules into the pores of VNU-23 with suitable acid–base interactions. According to this structural characterization, N-heterocyclic/VNU-23 materials,  $\text{Im}_{13.5}/\text{VNU-23}$ ,  $\text{mIm}_{9.2}/\text{VNU-23}$ , and  $\text{bIm}_{3.7}/\text{VNU-23}$ , exhibited high proton conductivity values under 85% RH and a moderately high temperature of 70 °C in comparison with parent  $\text{H}^+/\text{VNU-23}$ . Notably,  $\text{Im}_{13.5}/\text{VNU-23}$  reveals the highest proton conductivity in all N-heterocyclic/VNU-23 composites and reaches a value of  $1.58 \times 10^{-2} \text{ S cm}^{-1}$  (85% RH and 70 °C). To the best of our knowledge, the proton conductivity of  $\text{Im}_{13.5}/\text{VNU-23}$  betters that of several of the highest performing MOF materials at  $\leq 70$  °C and 85% RH. In addition,  $\text{Im}_{13.5}/\text{VNU-23}$  indicates the excellent durability and stability for proton conduction, which is retained without any reduction in performance of proton conductivity for at least 96 h. This finding is valuable for insight into the effect of N-heterocyclic molecules on the MOF structure for remarkably enhancing the proton conductivity, which is helpful for further synthetic procedures of new high proton conductivity composites.

## Conflicts of interest

The authors maintain that they have no conflict of interest for this communication.

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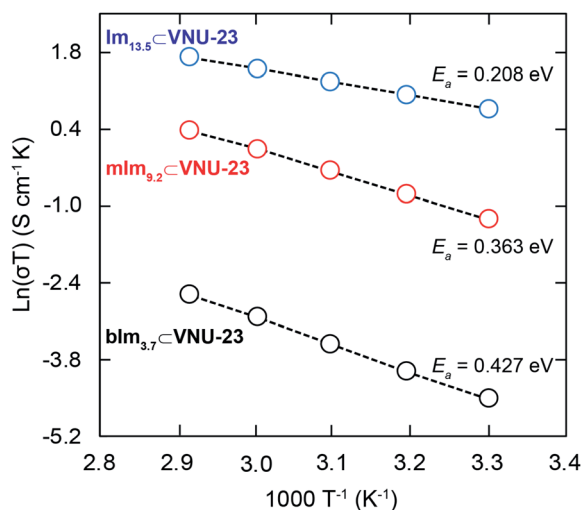


Fig. 7 Temperature-dependent proton conductivity measurements (30–70 °C) under 85% RH for  $\text{Im}_{13.5}/\text{VNU-23}$  (blue),  $\text{mIm}_{9.2}/\text{VNU-23}$  (red), and  $\text{bIm}_{3.7}/\text{VNU-23}$  (black).



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