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# DFT study on the electronic structure and optical properties of an Au-deposited $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> (001) surface

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The electronic structure and optical properties of gold clusters deposited on an  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> surface were studied by using density functional theory (DFT), with a special emphasis on the influence of Au cluster sizes. There is a strong interaction between Au clusters and the  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> surface, and the binding energy increases with an increase of Au cluster size. The Au atoms of the gold cluster are bonded to the iron atoms of the  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> surface for the Au/ $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> system, and the electrons transfer from the Au cluster to the  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> surface with the largest number of electrons transferred for 4Au/ $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>. The peaks of the refractive index, extinction coefficient and dielectric function induced by Au clusters appear in the visible range, which results in the enhanced optical absorption for the Au/ $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> system. The optical absorption intensifies with increasing Au cluster size in the visible range, showing a maximum value for 4Au/ $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>. Further increasing the Au cluster size above 4Au results in a decrease in absorption intensity. The results are in good agreement with those of the refractive index, extinction coefficient and dielectric function.

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## 1 Introduction

Long-term exposure to high concentrations of volatile organic compounds (VOCs) could cause serious damage to the lungs and induce various cancers.<sup>1–3</sup> In recent years, several studies have been reported on the removal of VOCs.<sup>4,5</sup> After years of development, some removal methods have been developed, such as incineration,<sup>6</sup> condensation,<sup>7</sup> adsorption and absorption<sup>8</sup> and so on. However, these methods have their own shortcomings, which limits their application in practice. For example, incineration and condensation have good cost performance only for medium and high VOC concentrations.<sup>9</sup> Adsorption and absorption only transfer VOCs to another medium, and do not destroy them. Furthermore, the adsorption part of the adsorbent could be clogged and filled by moist airflow. Herein, photo-catalytic oxidation has attracted increasing attention for its effective efficiency and low budget for low VOC concentrations.

Various photo-catalysts (TiO<sub>2</sub>, ZnS, CdS, SnO<sub>2</sub> and WO<sub>3</sub>) have been developed by the continuous efforts. However, most oxides of these semiconductors can only absorb UV light, which severely limits their practical applications. Although sulfides have a lower band gap and better stability than oxides, these

materials are very unstable in liquid media.<sup>10</sup> Compared with other photocatalytic materials,  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> possesses a lower band gap (2.2 eV) with a strong capacity of light absorption in the visible region. In addition,  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> is stable in most aqueous solutions (pH > 3) and is one of the cheapest semiconductor materials available.<sup>11,12</sup> Therefore, it has been investigated as a potential photocatalytic material in recent years. For example, Dong *et al.* discovered grain-like  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> nanostructures can efficiently degrade the congo red dyes.<sup>13</sup> Nevertheless, the photocatalytic activity of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> is limited by high position of valence, weak diffusion capacity of photo-generated holes and low conductivity.<sup>14,15</sup> In order to overcome these defects, a lot of research has been carried out, including surface modification,<sup>16,17</sup> photosensitization,<sup>18,19</sup> and doping with different materials,<sup>20,21</sup> *etc.*

Noble metal deposition<sup>22</sup> is an effective strategy to overcome the limitation of high recombination rate and poor electrical conductivity of electron-hole for  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>. The deposition of noble metals on the catalyst surface by local surface plasmon resonance (LSPR) makes the metal nanoparticles have a strong absorption effect on the photon energy.<sup>23</sup> Generally, the photocatalytic reaction is determined by three main energy transfer mechanisms, including the orbital leap of charge, the enhancement of the nearby electric field and the scattering mechanism. Liu *et al.*<sup>24</sup> synthesized porous  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> decorated by Au nanoparticles, which exhibits a much higher response in comparison to pure  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>. Li *et al.* found<sup>25</sup> that the Schottky barrier formed between precious metals and semiconductors

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effectively promotes the separation of light-induced electrons and holes, and the photocurrent is significantly enhanced on the visible-light catalytic oxidation of refractory organic compounds of Au/ $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>. Nevertheless, the degradation rate of organic compounds decreases with the increase of the loading amounts. The overloaded Au particles turns into the recombination center of photogenerated electrons and holes, and the generation of photocurrent is inhibited with the decrease of quantum yield. Lin *et al.*<sup>26</sup> successfully loaded Au and Pd on the surface of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> by the reduction method of polyvinyl alcohol, and the photocatalytic degrading VOCs efficiency of modified  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> could be significantly improved because of the large optical response range. Combined with photo-assisted growth, gold nanoparticles were deposited successfully on  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> nanocrystals through a cysteine-linked seed planting approach by Cao *et al.*,<sup>27</sup> and the size of the gold particles on  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> nanocrystals was regulated from 2 to 20 nm through a photo-induced seed mediation scheme. The Au deposited  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> nanocrystals not only exhibit efficient catalysis properties for 4-NP reduction but also have great efficiency for visible-light-driven photocatalytic dye degradation and water oxidation.

Metal clusters consist of several to hundreds of atoms have been popularized very much in recent years.<sup>28,29</sup> It is usually that metal nanoclusters possess a very small size that is comparable to the Fermi wavelength (the de Broglie wavelength of the conduction electrons present near the Fermi energy level) compared with nanoparticles. The size of metal clusters (Ag and Au) is usually smaller than 2 nm.<sup>28</sup> The metal nanoclusters are more like photonic supramolecules than the plasmonic nanoparticle analogue. For quantum confinement, the metal nanoclusters exhibit discrete molecular-like energy levels and a size-dependent LUMO–HOMO gap, meanwhile, they lose the related properties of the bulk ones, such as the SPR effect, which means the surface electrons in metal nanoclusters no longer oscillate together. Hence, the chemical and physical properties of the metal nanoclusters change to make the energy transfer more different and complicated from their larger counterparts.<sup>30</sup> For bulk Ag metal nanoparticles in the aqueous phase, the Ag<sup>+</sup>/Ag redox potential is reported to be +0.79 V (vs. NHE) but is –1.80 V (vs. NHE) for Ag nanoclusters.<sup>31</sup> Carolina *et al.*<sup>32</sup> investigated the structures and the growth processes of small Cu, Ag and Au clusters on the Fe-terminated (0001) surface of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> using DFT method. Hon *et al.*<sup>33</sup> studied Au nanoclusters supported on the  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> (0001) surface by the PBE+U calculations, and found that the periphery atoms of Au<sub>10</sub> particles become oxidized through the dissociation of O<sub>2</sub> at the metal-oxide interface. Nguyen *et al.*<sup>34</sup> studied the different properties of Au monomers and dimers on the defect-free and defective terminations of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> (0001) surface and their interaction with CO through DFT calculation. The results showed that the range of charge states of Au atoms is somewhere in between 0 and 1, besides, the most active site for CO oxidation is probable Au replacing a Fe in the Fe<sub>2</sub>O<sub>3</sub> surface. Pabisiak *et al.*<sup>35</sup> investigated the adsorption of small Au<sub>n</sub> nanocluster on the iron terminated Fe<sub>2</sub>O<sub>3</sub>(0001) surface and CO on the Au<sub>n</sub> cluster supported on the Fe<sub>2</sub>O<sub>3</sub>(0001) surface. They got the conclusion that the interaction of CO molecule with Au<sub>n</sub> cluster on the hematite surface is stronger

than that with pure Fe<sub>2</sub>O<sub>3</sub>(0001) surface. Overall, the photocatalytic activity of noble metal cluster deposition on  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> surface could be enhanced *via* facilitating the electron movement on the heteroboundary surface and hindering the recombination of photogenerated electron holes, and the amount and size of loaded metals cluster are of significance to the catalytic activity of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>.

Although Au cluster deposition on  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> (0001) surface have been studied, the microscopic mechanism of the influence of Au cluster size on  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> photocatalytic performance haven't been described in detail. Here, we focus on the electronic structures and optical properties of Au clusters deposited  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> surface by using first principle method based on density functional theory (DFT), with a special emphasis on the influences of Au cluster size. This study will provide an important theoretical basis for the application of metal-enhanced photocatalytic materials in industry.

## 2 Computational methods and models

### 2.1 Computational method

All calculations were carried out by using the CASTEP (Cambridge Sequential Total Energy Package) module in the MS (material studio) software package by Payne *et al.*<sup>36</sup> The exchange-correlation functional of the generalized gradient approximation (GGA), developed by Perdew and Wang (PW91),<sup>37</sup> A Monkhorst–Pack *k*-point sampling density of  $3 \times 3 \times 1$  and a plane wave cutoff energy of 340 eV were used for all calculations based the test results. Under these parameters, the lattice constant and band gap of iron oxide are close to the experimental values, and the system energy is relatively low, which was described in detail elsewhere.<sup>38</sup> The interactions between valence electrons and ionic core were represented with ultrasoft pseudopotentials.<sup>39</sup> The dipole correction with self-consistent

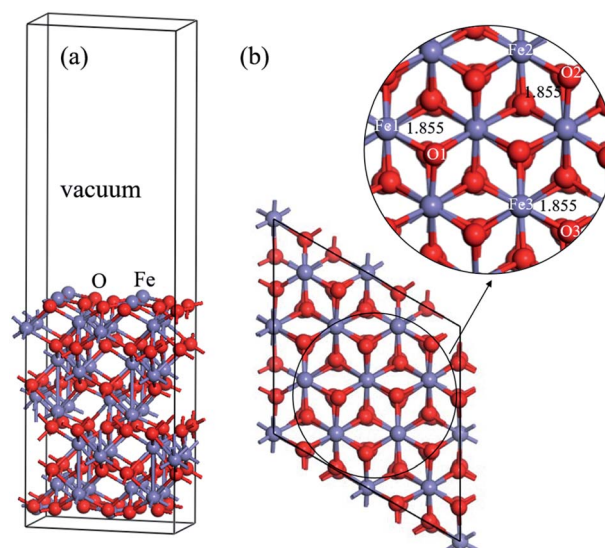


Fig. 1 Supercell model of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> (001) surface. (a) Main view; (b) top view. Numbers indicate the distance in Å.



was used in calculation. In order to decrease the error of calculated band gap, the calculations used the DFT+U method with  $U$  value of 3.5 eV based the test. Valence electrons configuration considered in this study included O  $2s^2 2p^4$ , Fe  $3d^6 4s^2$  and Au  $5d^{10} 6s^1$  states. The convergence tolerances for geometry optimization calculations were set to the maximum displacement of 0.002 Å, the maximum force of 0.05 eV Å<sup>-1</sup>, the maximum energy change of  $2.0 \times 10^{-5}$  eV per atom and the maximum stress of 0.1 GPa, and the self-consistent field (SCF) convergence tolerance was set to  $2.0 \times 10^{-6}$  eV per atom.

## 2.2 Computational model

Hematite ( $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>) possesses a typical rhombohedral structure with a space group of  $R\bar{3}c$ . Each Fe atom coordinates with

adjacent six O atoms, and each O atom coordinates with four Fe atoms, forming an corundum crystal structure, which has been described in detail elsewhere.<sup>40</sup> The lattice constants of the  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> bulk are  $a = b = 5.088$  Å, and  $c = 14.138$  Å, which are close to the experimental results ( $a = b = 5.0355$  Å, and  $c = 13.747$  Å). In addition, the bandgap of the Fe<sub>2</sub>O<sub>3</sub> bulk is 2.265 eV, which is in accord with the experimental value (1.9–2.2 eV).<sup>41</sup> These results were described in detail by our group.<sup>42</sup>  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> (001) surface was cut from the bulk with the optimum unit cell volume. The adjacent mirror images of the slab can be separated under a vacuum environment, the thickness of vacuum layer in the vertical direction is set as 15 Å, and the slab thickness was selected as 18 layers by testing the surface energies of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> (001). The  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> surface was modeled using  $2 \times 2 \times 1$

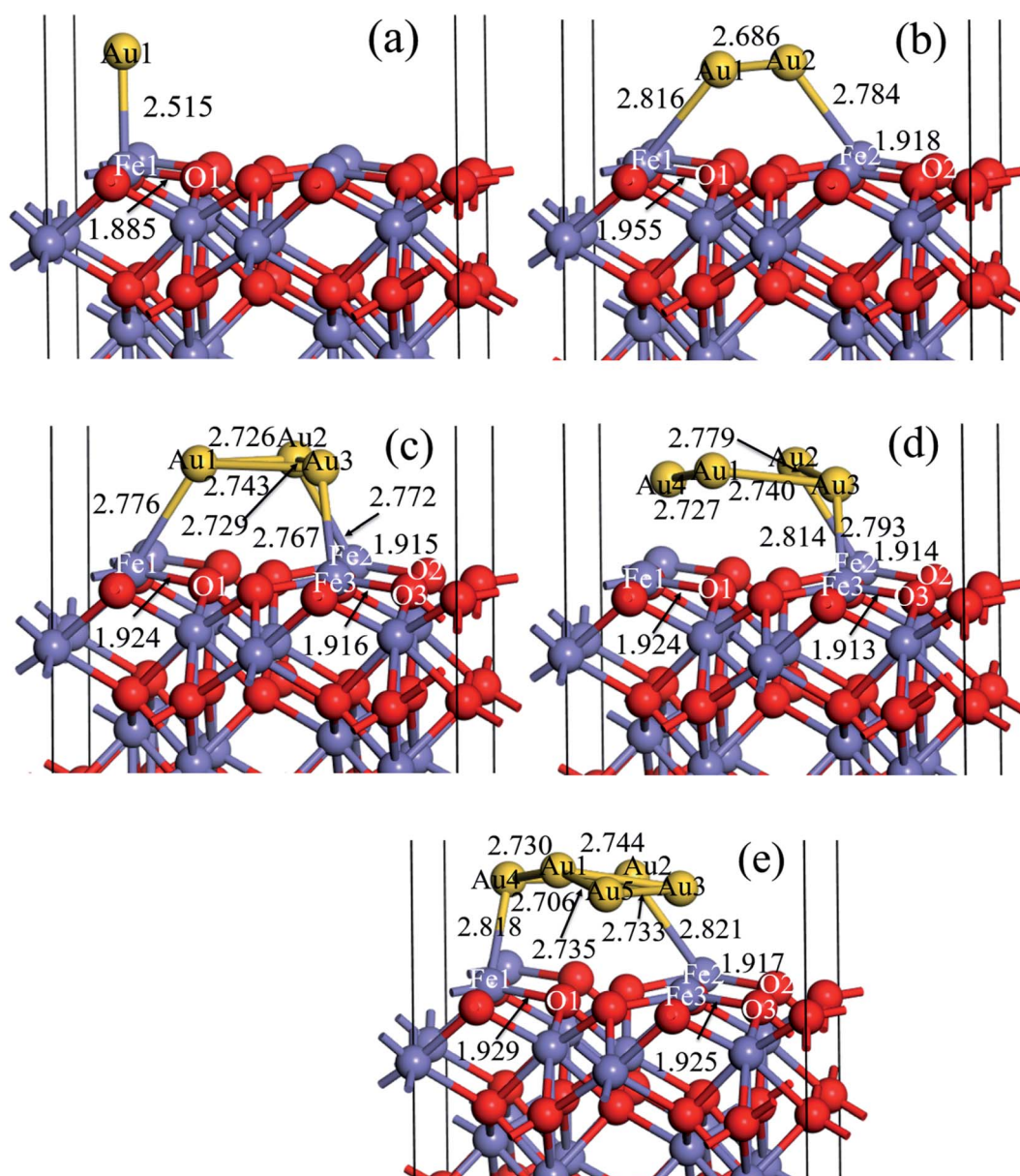


Fig. 2 Equilibrium configuration of (a) 1Au/ $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> (001) surface, (b) 2Au/ $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> (001) surface (c) 3Au/ $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> (001) surface surface, (d) 4Au/ $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> (001) surface, (e) 5Au/ $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> (001) surface. Numbers indicate the distance in Å.



supercell geometries for better research of Au particle deposited  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> surface. The surface energies of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> (001) with varying slab thicknesses were tested to determine the slab size and finally established the most stable slab model, as shown in Fig. 1.

## 3 Results and discussion

### 3.1 Configurations of Au/ $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> (001) surface

In order to determine the stable configuration of Au deposited on  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> surface, we studied the deposition of one Au atom on  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> surface with different sites, including the top deposition of Au atom on iron of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> surface, top deposition on oxygen, and vacancy deposition of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> surface, and calculated their binding energies. The results show that the top deposition of Au atom on iron of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> surface is the most stable. As a result, we built the deposition models of multiple Au atoms by putting Au atoms at the top of iron atom of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> surface. Fig. 2 shows the configurations of Au/ $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> surface with different Au atoms. For single Au atom deposited on  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> surface, Au1 atom is bonded to Fe1 atoms of the  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> surface, and the bond length of Au1–Fe1 is 2.515 Å (Fig. 2(a)). For 2Au/ $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>, on the one hand, Au1 atom is bonded to Au2 atom with a bond length of 2.686 Å; on the other hand, Au1 and Au2 atoms are bonded to Fe1 and Fe2 atoms of the  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> surface, respectively, and the bond lengths of Au1–Fe1 and Au2–Fe2 are 2.816 Å and 2.784 Å, respectively (Fig. 2(b)). Nguyen *et al.*<sup>34</sup> studied the deposition of Au monomers and dimers on the Fe<sub>2</sub>O<sub>3</sub> (0001) surface. They found that Au atom is bonded to the iron atom of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> surface for monomers, and the bond length of Au–Fe is 2.49 Å. As for Au dimer, Au1 atom is bonded to Au2 atom with a bond length of 2.56 Å. In addition, Au1 and Au2 atoms are bonded to Fe and O atoms of the  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> surface, respectively, and the bond length of Au1–Fe is 2.66 Å. Pabisiak *et al.*<sup>35</sup> also investigated small Au clusters structures supported on iron termination of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>(0001) surface. They found that Au atom for Au monomer is bonded to an iron of surface with a bond length of 2.52 Å. The length of Au1–Au2 bond for Au dimer is 2.59 Å, and Au1 and Au2 atoms are bonded to iron and oxygen atoms of the  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> surface, respectively with the Au1–Fe bond of 2.67 Å and Au2–O bond of 2.17 Å. For the deposition of a gold atom on  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> surface, our result is similar with those of Nguyen and Pabisiak *et al.* Au is bonded to an iron of the surface with a bond length of about 2.5 Å. However, there are some differences for dimers. Two Au atoms are bonded to two Fe atoms of the surface in our study, while in Nguyen and Pabisiak's study, two Au atoms are bonded to Fe and O atoms of the surface, respectively. The reason may be different modeling methods. Two Au atoms are simultaneously placed on the surface of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> in our modeling, while in Nguyen and Pabisiak's modeling, the Au dimer was constructed with the second Au atom positioned on the neighboring site of the first Au atom which was already adsorbed on  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> surface. For the difference, we will further study in the future. For 3Au/ $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>, three Au atoms are bonded to each other, forming a triangular cluster structure, meanwhile, they are also bonded to three iron atoms of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> surface (Fe1, Fe2 and Fe3),

respectively. The lengths of newly formed Au1–Au2, Au1–Au3, Au2–Au3, Au1–Fe1, Au2–Fe2 and Au3–Fe3 bonds are 2.726 Å, 2.743 Å, 2.729 Å, 2.776 Å, 2.772 Å and 2.767 Å, respectively (Fig. 2(c)). For 4Au/ $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>, the Au1 atom is bonded to Au3 and Au4 atoms, and Au3 atom is bonded to Au2 atom besides Au1 atom. In addition, Au3 and Au2 are also bonded to Fe3 and Fe2 of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> surface, respectively. The lengths of newly formed Au4–Au1, Au1–Au3, Au3–Au2, Au2–Fe2 and Au3–Fe3 bonds are 2.717 Å, 2.740 Å, 2.779 Å, 2.814 Å and 2.793 Å, respectively (Fig. 2(d)). As for 5Au/ $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>, Au1, Au3 and Au5 atoms form a triangular structure. Au4 atom is bonded to Au1 and Au2 atoms, and Au4 and Au2 atoms are bonded to Fe1 and Fe2, respectively. All bond lengths are shown in Fig. 2(e).

It is found from Fig. 2 that Au atoms all interact with iron atoms of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> surface. Kiejna *et al.*<sup>43</sup> investigated the structure and electronic properties of clean and Au or Pd covered  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> (0001) surface with spin-polarized density functional theory (DFT) and the DFT+U method. The results showed that both Au and Pd atoms bind strongly to hematite surfaces and induce large changes in their geometry. For the iron terminated surface, the binding energy per adatom increases with Au and Pd coverage, whereas the opposite trend is observed for the oxygen terminated one, which is consistent with our Au/ $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> configurations. In addition, it is also observed that the bond lengths of Au–Fe for Fig. 2(a) and (c) are slightly smaller than that for Fig. 2(b), (d) and (e), indicating that the interactions between Au and Fe of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> surface of Fig. 2(a) and (c) may be stronger than those of other models. However, the bonding conditions are different in Au/ $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> systems with different number of gold atoms, resulting in their different stability. In addition to the interaction between gold and gold, there is an Au–Fe bond for 1Au/ $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>, two Au–Fe bonds for 2Au/ $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>, and three Au–Fe bonds for 3Au/ $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>, that is, the number of Au–Fe bonds increases with increasing the number of gold atoms. However, the number of Au–Fe bonds not only does not increase, but decreases when the number of gold atoms increases to four and five. There are two Au–Fe bonds for 4Au/ $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> and 5Au/ $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>.

To assure the strength of interaction between Au and  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> surface in different systems, the binding energies of Au/ $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> with different number of gold atoms are calculated (Table 1). The binding energies of Au particle deposited on  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> surfaces were calculated as:

$$E_{\text{bin}} = E_{\text{Au/surface}} - nE_{\text{Au}} - E_{\text{surface}} \quad (1)$$

**Table 1** Binding energies of Au/ $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> (001) surface with different Au amounts

Configurations	Energy (kJ mol <sup>−1</sup> )
Fig. 2(a)	−106.08
Fig. 2(b)	−266.17
Fig. 2(c)	−511.13
Fig. 2(d)	−684.72
Fig. 2(e)	−881.46



where  $E_{\text{bin}}$  is the binding energy,  $E_{\text{Au/surface}}$  is the energy of  $\alpha\text{-Fe}_2\text{O}_3$  slab with Au particle,  $n$  is the number of Au atoms,  $E_{\text{Au}}$  is the energy of an Au calculated in a cubic cell,  $E_{\text{surface}}$  is the energy of hematite slab. The binding energies of  $1\text{Au}/\alpha\text{-Fe}_2\text{O}_3$ ,  $2\text{Au}/\alpha\text{-Fe}_2\text{O}_3$ ,  $3\text{Au}/\alpha\text{-Fe}_2\text{O}_3$ ,  $4\text{Au}/\alpha\text{-Fe}_2\text{O}_3$  and  $5\text{Au}/\alpha\text{-Fe}_2\text{O}_3$  are  $-106.08\text{ kJ mol}^{-1}$ ,  $-266.17\text{ kJ mol}^{-1}$ ,  $-511.13\text{ kJ mol}^{-1}$ ,  $-684.72\text{ kJ mol}^{-1}$  and  $-881.46\text{ kJ mol}^{-1}$ , respectively. The negative sign represents exothermic reaction. It is observed that, the binding energy of  $\text{Au}/\alpha\text{-Fe}_2\text{O}_3$  increases from  $106.08\text{ kJ mol}^{-1}$  to  $881.46\text{ kJ mol}^{-1}$  with increasing Au atom concentration. However, the larger binding energy does not mean that the interaction between Au and  $\alpha\text{-Fe}_2\text{O}_3$  surface is stronger. The values of calculated binding energy include not only the interaction between Au and  $\alpha\text{-Fe}_2\text{O}_3$  surface, but also that between Au and Au. The binding energy of  $5\text{Au}/\alpha\text{-Fe}_2\text{O}_3$  is the largest, which may be due to the most Au–Au bonds.

The Mulliken bond population can reflect the strength of covalent bonds, the larger the bond population is, the stronger the covalent bond is. Tables 2–4 show the Mulliken bond populations of Fe–Au and Au–Au bonds for  $3\text{Au}/\alpha\text{-Fe}_2\text{O}_3$ ,  $4\text{Au}/\alpha\text{-Fe}_2\text{O}_3$  and  $5\text{Au}/\alpha\text{-Fe}_2\text{O}_3$ , respectively. It is observed that the populations of three Fe–Au bonds of  $3\text{Au}/\alpha\text{-Fe}_2\text{O}_3$ ,  $4\text{Au}/\alpha\text{-Fe}_2\text{O}_3$  and  $5\text{Au}/\alpha\text{-Fe}_2\text{O}_3$  are similar, however, populations of three Fe–Au bonds (Fe1–Au1, Fe2–Au2) of  $3\text{Au}/\alpha\text{-Fe}_2\text{O}_3$  are largest (0.14), and followed by two bonds (Fe1–Au4 and Fe2–Au2) of  $5\text{Au}/\alpha\text{-Fe}_2\text{O}_3$  (0.13) and two bonds (Fe2–Au2 and Fe3–Au3) of  $4\text{Au}/\alpha\text{-Fe}_2\text{O}_3$  (0.12) in order, showing the strength of Fe–Au bond for  $3\text{Au}/\alpha\text{-Fe}_2\text{O}_3$  is slightly larger compared to those of  $4\text{Au}/\alpha\text{-Fe}_2\text{O}_3$  and  $5\text{Au}/\alpha\text{-Fe}_2\text{O}_3$ . In addition, it is also seen that Mulliken bond populations of Au–Au in all models are much larger than those of Fe–Au, suggesting that the interaction between Au and Au is much stronger than that between Fe and Au. Therefore, the

**Table 4** The Mulliken bond population of the  $5\text{Au}/\alpha\text{-Fe}_2\text{O}_3$  (001) surface model

Bond types	Population	Bond length (Å)
Fe1–Au4	0.13	2.818
Fe2–Au2	0.13	2.821
Au2–Au4	0.37	2.730
Au1–Au4	0.34	2.706
Au1–Au3	0.25	2.744
Au1–Au5	0.30	2.735
Au3–Au5	0.41	2.733

binding energy is in increasing order of  $1\text{Au}/\alpha\text{-Fe}_2\text{O}_3$ ,  $2\text{Au}/\alpha\text{-Fe}_2\text{O}_3$ ,  $3\text{Au}/\alpha\text{-Fe}_2\text{O}_3$ ,  $4\text{Au}/\alpha\text{-Fe}_2\text{O}_3$  and  $5\text{Au}/\alpha\text{-Fe}_2\text{O}_3$  due to the increase of Au–Au bond (Table 1).

### 3.2 Electronic structures

The energy band and corresponding PDOS of up-spin and down-spin of pure  $\alpha\text{-Fe}_2\text{O}_3$  surface was studied previously.<sup>38</sup> The band gap of pure  $\alpha\text{-Fe}_2\text{O}_3$  surface depends on the top valence band of the spin-up band and the bottom of the conduction band of the spin-down band, and the value is 2.0 eV. The spin-up band is divided into three groups. The group between  $-9\text{ eV}$  and  $-5\text{ eV}$  is mainly from Fe 3d and O 2p, only with few contributions from Fe 4s, Fe 4p and O 2s. The group between  $-5\text{ eV}$  and  $0\text{ eV}$  is from O 2p, Fe 3d with a few contributions from Fe 4s and Fe 4p. The group in the conduction band is mainly from Fe 4p, Fe 4s, O 2p and O 2s. For the spin-down band, there is one group in the valence band, which is mainly from O 2p and Fe 3d with a few contributions from Fe 4p and Fe 4s. The band between  $2\text{ eV}$  and  $5\text{ eV}$  in the conduction band is mainly Fe 3d and O 2p with few contributions from Fe 4p, O 2s and Fe 4s, while that in the range of  $5\text{ eV}$  to  $8\text{ eV}$  is mainly from Fe 4p and Fe 4s with few contributions from O 2p and O 2s.

Fig. 3 shows the energy band and PDOS of spin-up (a) and spin-down (b) for  $1\text{Au}/\alpha\text{-Fe}_2\text{O}_3$ , spin-up (c) and spin-down (d) for  $2\text{Au}/\alpha\text{-Fe}_2\text{O}_3$  surface, spin-up (e) and spin-down (f) for  $3\text{Au}/\alpha\text{-Fe}_2\text{O}_3$  surface, spin-up (g) and spin-down (h) for  $4\text{Au}/\alpha\text{-Fe}_2\text{O}_3$  surface, spin-up (i) and spin-down (j) for  $5\text{Au}/\alpha\text{-Fe}_2\text{O}_3$  surface. For up-spin band and DOS, it is observed that the energy structure and corresponding DOS of the valence band with small Au cluster ( $1\text{Au}/\alpha\text{-Fe}_2\text{O}_3$ ) are similar with that of pure  $\alpha\text{-Fe}_2\text{O}_3$ .<sup>38</sup> There are two groups for the valence band of  $1\text{Au}/\alpha\text{-Fe}_2\text{O}_3$ . The group between  $-6$  and  $0\text{ eV}$  is mainly composes of O 2p and Fe 3d with few contribution from Fe 4s and Fe 4p, while that between  $-9$  and  $-5\text{ eV}$  is mainly from Fe 3d and O 2p. The two groups are connected to each other with increasing the size of Au cluster (from 2Au to 5Au), and the larger the Au cluster, the more the two groups overlap (Fig. 3(a), (c), (e), (g) and (i)). Compared to pure  $\alpha\text{-Fe}_2\text{O}_3$ ,<sup>38</sup>  $\text{Au}/\alpha\text{-Fe}_2\text{O}_3$  systems present the different energy structure for the conduction band. A new energy level at  $0.30\text{ eV}$  for  $1\text{Au}/\alpha\text{-Fe}_2\text{O}_3$  appears near the Fermi level, which is mainly from O 2p, Fe 3d, Au 6s and Au 6p. The number of new energy levels increases with increasing the size of Au cluster. There are two new energy levels for  $2\text{Au}/\alpha\text{-Fe}_2\text{O}_3$

**Table 2** The Mulliken bond population of the  $3\text{Au}/\alpha\text{-Fe}_2\text{O}_3$  (001) surface model

Bond types	Population	Bond length (Å)
Fe1–Au1	0.14	2.776
Fe2–Au2	0.14	2.772
Fe3–Au3	0.14	2.767
Au1–Au2	0.35	2.726
Au1–Au3	0.34	2.743
Au2–Au3	0.34	2.729

**Table 3** The Mulliken bond population of the  $4\text{Au}/\alpha\text{-Fe}_2\text{O}_3$  (001) surface model

Bond types	Population	Bond length (Å)
Fe2–Au2	0.12	2.814
Fe3–Au3	0.12	2.793
Au1–Au4	0.35	2.727
Au1–Au3	0.28	2.740
Au2–Au3	0.27	2.779
Au1–Au2	0.26	2.804
Au2–Au4	0.26	2.804



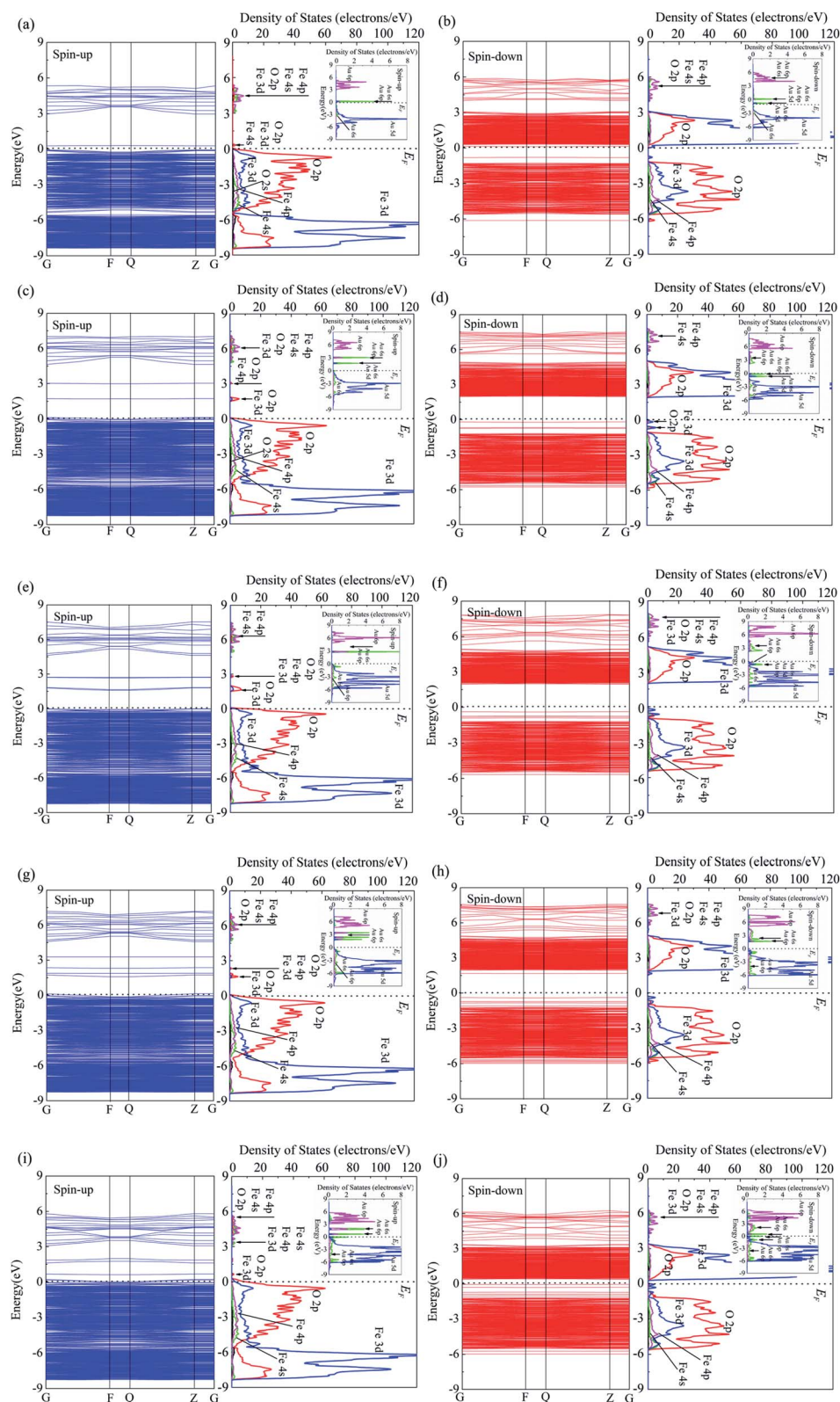


Fig. 3 Energy band and PDOS of spin-up (a) and spin-down (b) for 1Au/ $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> (001) surface, spin-up (c) and spin-down (d) for 2Au/ $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> (001) surface, spin-up (e) and spin-down (f) for 3Au/ $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> (001) surface, spin-up (g) and spin-down (h) for 4Au/ $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> (001) surface and spin-up (i) and spin-down (j) for 5Au/ $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> (001) surface.





and 3Au/ $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>, and they move away from the Fermi level relative to 1Au/ $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> (Fig. 3(c) and (e)). The energy level at 1.72 eV is mainly from O 2p and Fe 3d, while that at 3.01 eV is mainly from O 2p and Fe 4p. For 4Au/ $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>, there are four new energy levels about 1.71, 1.90, 2.35 and 3.27 eV respectively. It is also found that the DOS peaks about 3.27 eV and 2.35 eV are weak, and the DOS curves corresponding to energy levels at about 1.71 eV and 1.90 eV are connected to each other to form a DOS peak, which is from O 2p, Fe 3d, Au 6s and Au 6p (Fig. 3(g)). As for 5Au/ $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>, there are also four new energy levels, however, these energy levels, including all energy levels of the conduction band move toward to Fermi level relative to those of 4Au/ $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>. The DOS curve corresponding to the energy level closest to the Fermi level overlaps with DOS curve of the valence band. The other three energy levels are located at 0.71, 1.63, 1.86 eV, respectively, and their corresponding DOSs are very weak. Changeable rule of DOS curve of Au cluster with different sizes is similar with that of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> matrix. It is found that the spin-up DOS of Au cluster for the valence band is divided into two groups for 1Au/ $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>. The group between -1 and 0 eV is mainly from Au 5d with few contribution of Au 6s, and that between -8 and -1 eV is attributed to Au 6s. DOS curve of the conduction band between 3 to 6 eV for Au cluster is mainly from Au 6p with few contribution of Au 6s, and that with large localization near Fermi level is attributed to Au 6s, Au 6p

and Au 5d (illustration in Fig. 3(a)). The two groups of the valence band are connected to each other with an increase of Au cluster size from 2Au to 5Au. There are two DOS peaks of Au cluster with large localization for 2Au/ $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> and 3Au/ $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> (illustration in Fig. 3(c) and (e)), three DOS peaks for 4Au/ $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> and 5Au/ $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> (illustration in Fig. 3(g) and (i)), which are from Au 6s, Au 6p and Au 5d.

For down-spin band, it is seen that two groups of the conduction band for 1Au/ $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> move toward Fermi level relative to that of pure  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>.<sup>38</sup> However, they move away from the Fermi level with increasing the size of Au cluster from 2Au to 4Au, and the interval between two groups decreases. When 5Au are deposited in  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> surface, the energy levels of the conduction band are very close to Fermi level. It is also observed that DOS curves change little with increasing the size of Au cluster from 1Au to 5Au, although there is a large shift for the energy levels of the conduction band. Unlike up-spin energy structure, new energy levels appears at the top of the valence band of down-spin band for Au/ $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> systems. There is a new energy level about -0.82 eV for 1Au/ $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>, which is from Fe 3d and Fe 4p. The number of new energy level increases with increasing the size of Au cluster from 2Au to 5Au. There are two new energy levels for 2Au/ $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> at the top of the valence band, three for 3Au/ $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>, four for 4Au/ $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> and five for 5Au/ $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>. Two new energy level for 2Au/ $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> corresponds to two

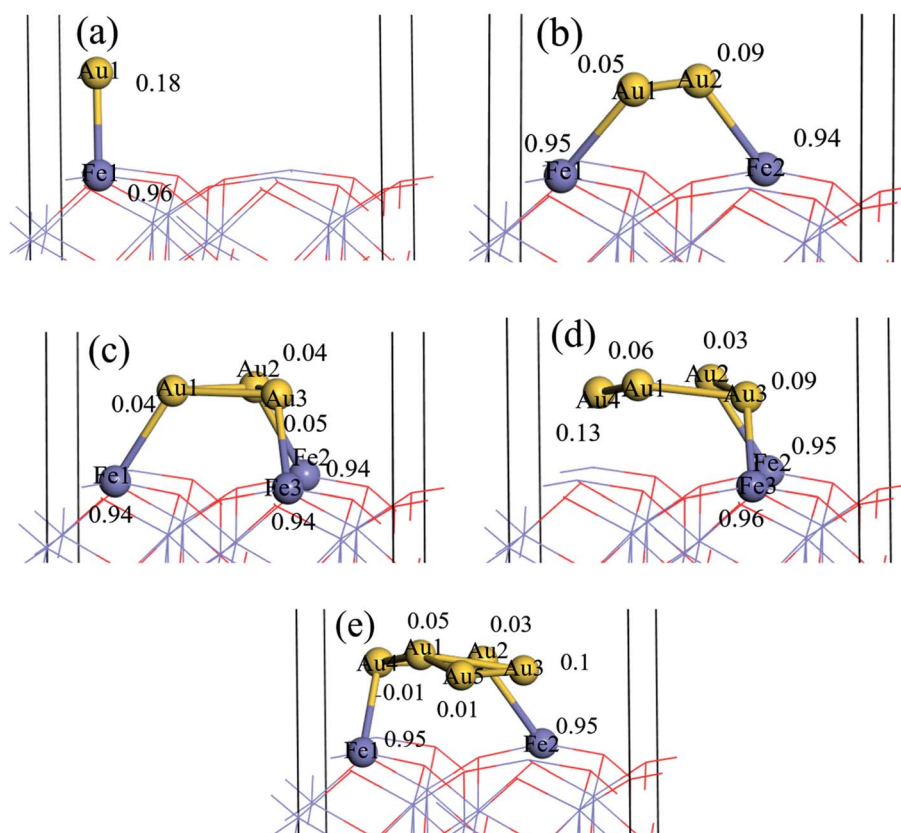


Fig. 4 Mulliken atomic charge population of (a) 1Au/ $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> (001) surface, (b) 2Au/ $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> (001) surface (c) 3Au/ $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> (001) surface surface, (d) 4Au/ $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> (001) surface, (e) 5Au/ $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> (001) surface.

DOS peaks, which are from Fe 3d, O 2p, Fe 4s and Fe 4p. Three new energy levels for 3Au/ $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> only exhibit two DOS peaks because the two energy levels far from the Fermi level are very close, so they overlap each other to form a peak corresponding to Fe 3d and O 2p. The peak closest to Fermi level also is from Fe 3d and O 2p. Similarly, due to the overlap of DOS curves, four energy levels for 4Au/ $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> and five energy levels for 5Au/ $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> exhibit two DOS peaks and three DOS peaks, respectively. Two DOS peaks for 4Au/ $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> are from Fe 3d and O 2p, and three DOS peaks for 5Au/ $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> are from Fe 3d and Fe 4s.

The down-spin DOS of Au cluster for the conduction band is divided into two groups for 1Au/ $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>. The group between 0 and 3 eV is very weak, and the group between 3 and 6 eV is mainly from Au 6p with few contribution of Au 6s. DOS curve between 3 and 6 eV for 1Au/ $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> shifts to high energy with increasing Au cluster size from 2Au to 3Au, however, it shifts to low energy when the Au cluster size continues to increase from 4Au to 5Au. The DOS curve between 0 and 3 eV corresponds to Au 6s, Au 6p and Au 5d, which becomes stronger with an increase of Au cluster size from 2Au to 5Au. The down-spin DOS curve of Au cluster for the valence band can also be divided into two parts for 1Au/ $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>. The part between -7 and -2 eV is mainly from Au 5d with few contribution of Au 6s. There are two sharp peaks in the range of -2 to 0 eV. One near Fermi level corresponds Au 6s and Au 6p, and the other is from Au 6s and Au 5d. The two peaks are close to each other with an increase of Au cluster size from 2Au to 4Au, and become weak. However, the peaks become stronger when five Au atoms are deposited on  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> surface.

Fig. 4 shows the atomic Mulliken charge population of Au/ $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> systems. It is seen that from Fig. 4(a) the charge of Au is 0.18e when 1Au is deposited on  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> surface, suggesting that the electrons transfer from Au to  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> surface. After 2Au depositing, the charges of Au1 and Au2 are 0.05e and 0.09e, respectively, and the total charge is 0.14e (Fig. 4(b)). After 3Au depositing, the charges of Au1, Au2 and Au3 are 0.04e, 0.04e and 0.05e, respectively, and the total charge is 0.13e (Fig. 4(c)). As shown in Fig. 4(d), the charges of 4Au are 0.06e, 0.03e, 0.09e, and 0.13e, respectively, and the total charge is 0.31e. As for 5Au/ $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>, it is observed that the charges of 5Au are 0.05e, 0.03e, 0.1e, -0.01e, and 0.01e respectively with the total charge of 0.18e (Fig. 4(e)). According to above analysis, Au atoms lose electrons after Au clusters depositing on  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> surface, suggesting the electrons transfer from Au

clusters to the  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> surface for Au/ $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> system. And the total number of the charges for Au cluster decreases with increasing the Au cluster size from 1Au to 3Au (from 0.18e to 0.13e), and increases again when the Au cluster increases to 4Au (0.31e). The total charges of Au cluster with 5Au is 0.18e, which is the same with that of 1Au. Therefore, for Au/ $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> systems, the number of charge transfer between Au cluster and  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> surface is the largest for 4Au/ $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> (0.31e), which suggests the interaction between Au clusters and Fe<sub>2</sub>O<sub>3</sub> surface for 4Au/ $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> is the strongest.

### 3.3 Optical properties

Fig. 5 shows real parts (a) and imaginary parts (b) of complex refractive index for pure and Au-deposited  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> (001) surface. The complex refractive index can be expressed as follows:

$$N = n + ik \quad (2)$$

The real ( $n$ ) and imaginary ( $k$ ) parts of complex refractive index are the refractive index and extinction coefficient, respectively. It is seen from Fig. 5 that there is a sharp peak of the refractive index with a value of 2.35 in the 3.37 eV and a gentle peak of extinction coefficient with a value of 1.31 in the 6.13 eV for pure  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> surface. When Au clusters are deposited on the  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> surface, the curves of refractive index and extinction coefficient change a lot. Notably, there are two peaks for the refractive index and extinction coefficient in the range of 0 to 10 eV. It is obvious that the first peaks for the refractive index at about 1.15 eV and extinction coefficient at about 6.08 eV are induced by Au cluster. The peak value of the refractive index for 2Au/ $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>, 3Au/ $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>, 4Au/ $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> and 5Au/ $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> are 2.18, 2.47, 2.60 and 2.29, respectively. The result shows that the intensity of the first peak increases with increasing Au cluster size from 2Au to 4Au, and decreases with further increasing Au cluster size above 4Au, exhibiting a maximum value with 4Au. The changeable rule of the extinction coefficient with the increase of Au cluster size is similar with that of the refractive index. The first peak of the extinction coefficient intensifies with an increase of Au cluster size up to 4Au, but weakens with increasing further Au cluster above 4Au.

Fig. 6 shows real parts (a) and imaginary parts (b) of complex dielectric function for pure and Au-deposited  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> surface.

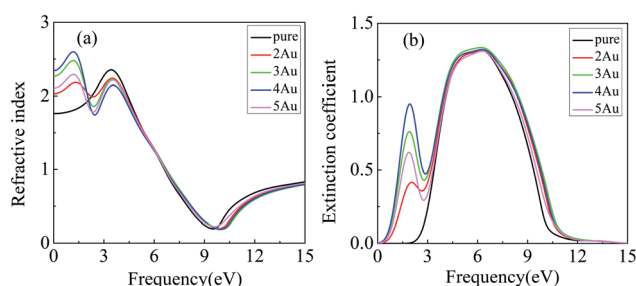


Fig. 5 Real parts (a) and imaginary parts (b) of complex refractive index for pure and Au-deposited  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> (001) surface.

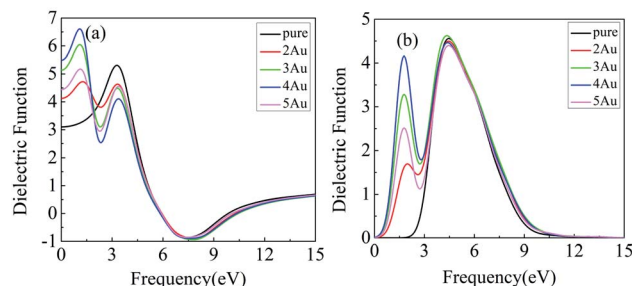


Fig. 6 Real parts (a) and imaginary parts (b) of complex dielectric function for pure and Au-deposited  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> (001) surface.





The complex dielectric function was using the following equation:

$$\varepsilon = \varepsilon_1 + i\varepsilon_2 \quad (3)$$

where  $\varepsilon_1$  and  $\varepsilon_2$  are the real and imaginary parts of complex dielectric function, respectively. The relation between the complex refractive index and the complex dielectric function can be expressed as follows

$$\varepsilon = N^2 \quad (4)$$

Therefore, the relation between the real part ( $\varepsilon_1$ ) and imaginary part ( $\varepsilon_2$ ) of complex dielectric function ( $\varepsilon$ ) and refractive index ( $n$ ) and extinction coefficient ( $k$ ) is:

$$\varepsilon_1 = n^2 - k^2 \quad (5)$$

$$\varepsilon_2 = 2nk \quad (6)$$

We can infer from formulas (4) and (5) that the real part of dielectric function increases with an increase of refractive index and a decrease of extinction coefficient, and the imaginary part increases with increasing the refractive index and extinction coefficient. It is observed from Fig. 6 that the real part and imaginary part of the complex dielectric function are similar to the change of refractive index and extinction coefficient for pure and Au-deposited  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>, respectively. The real part of the dielectric function at the frequency of 0 eV corresponds to the static dielectric constant. It is found that the static dielectric constants of 2Au-deposited, 3Au-deposited, 4Au-deposited and 5Au-deposited  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> surface are 4.14, 5.14, 5.49 and 4.43, respectively, which is far larger than that of pure  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> (3.11), suggesting that the deposition of Au clusters lead to an increase of static dielectric constant of the  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> surface. Furthermore, it is also observed that the Au cluster size has a great influence on the static dielectric constant of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>. The static dielectric constant increases with increasing Au cluster size up to 4Au, but decreases with increasing further the Au cluster size above 4Au. There is a sharp peak of the real part at about 3.26 eV and a sharp peak of the imaginary part at about 5.2 eV for pure  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>. After the deposition of Au clusters, there are two peaks for the real part and imaginary part of the complex dielectric function. Obviously, the peak at about 3.26 eV for the real part corresponds to the peak of pure  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>, while the peak at 1.12 eV is induced by Au clusters. It is also found from Fig. 6(a) that the peak from  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> matrix is weakened due to the influence of Au clusters, and that of 4Au/ $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> weakens the most, while the peak induced by Au clusters intensities with increasing Au cluster size up to 4Au, but weaken with increasing further Au cluster size beyond 4Au, showing a maximum value for 4Au (peak value: 2Au-Fe<sub>2</sub>O<sub>3</sub>, 4.72; 3Au-Fe<sub>2</sub>O<sub>3</sub>, 6.04; 4Au-Fe<sub>2</sub>O<sub>3</sub>, 6.62; 5Au-Fe<sub>2</sub>O<sub>3</sub>, 5.21). The peak at about 4.58 eV for the imaginary part corresponds to that of pure  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>, while the peak at 1.85 eV is induced by Au clusters (Fig. 6(b)). It is found that the peak from  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> matrix has not much change with different Au cluster size, while the peak induced by Au cluster intensities first, and then weakens with an

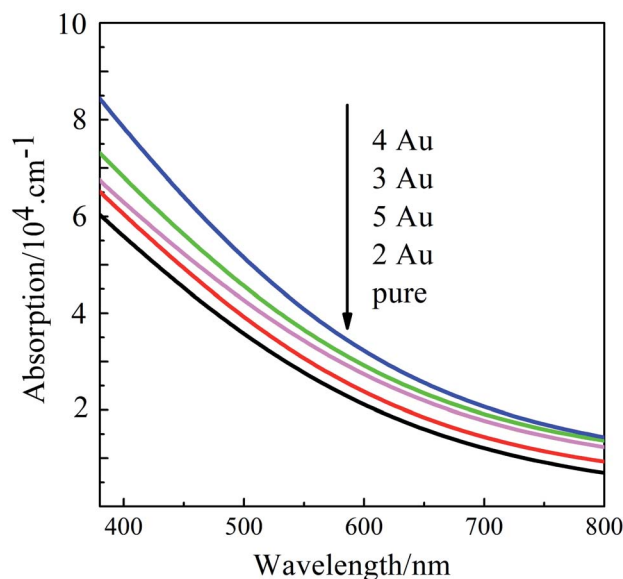


Fig. 7 Optical absorption of pure and Au-deposited  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> (001) surface.

increase of Au cluster size, showing a maximum for 4Au (peak value: 2Au-Fe<sub>2</sub>O<sub>3</sub>, 1.69; 3Au-Fe<sub>2</sub>O<sub>3</sub>, 3.29; 4Au-Fe<sub>2</sub>O<sub>3</sub>, 4.16; 5Au-Fe<sub>2</sub>O<sub>3</sub>, 2.53), which is consistent with those of the real part and imaginary part of complex refractive index.

The optical absorption of pure and Au-deposited  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> surface are shown in Fig. 7. It is found that the pure  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> surface shows a weak absorption in the range of 380 to 800 nm, however, the optical absorption is obviously enhanced after Au clusters deposition. The intensity of optical absorption increases with increasing the Au cluster size up to 4Au, but weakens with increasing further Au cluster size above 4Au, showing a maximum value for 4Au/ $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>. The optical absorption coefficient ( $\eta$ ) is related to the extinction coefficient ( $k$ ) by:

$$\eta = 2k\omega/c \quad (7)$$

where  $c$  is the speed of light, and  $\omega$  is the frequency. It is seen that the optical absorption is proportional to  $k$ . According to Fig. 7, the peaks of the extinction coefficient induced by Au clusters appear between 1.5 eV and 3.2 eV (corresponding wavelength: 380–800 nm) for Au-deposited  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> system, and the peak intensity increases with increasing Au cluster size up to 4Au, but decreases above 4Au. The changeable rule with the increase of Au cluster size is well agreement with the result of optical absorption in the visible range. Therefore, the deposition of Au clusters can effectively improve optical absorption of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> in the visible range, and the size of the Au cluster affects the intensity of optical absorption.

## 4 Conclusion

Based on the density functional theory (DFT), we studied the electronic structure and optical property of pure and Au-deposited  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> surface, and focused on the influence of Au cluster size. Au atoms are prone to interact with iron atoms of  $\alpha$ -



Fe<sub>2</sub>O<sub>3</sub> surface, which results in a great change of the structure of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> surface. The length of Fe–O bond of Au/ $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> surface is obviously larger compared with those of pure  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> surface. The bandgap and semiconducting properties change after Au cluster depositing. The energy level in the conduction band with the corresponding PDOS curve shifts to Fermi level, and the part near to Fermi level is split, and. The series of new energy level for Au/ $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> appears near the Fermi level, which is mainly from O 2p, Fe 3d, Au 6s and Au 6p. In addition, Au atoms lose electrons after Au clusters are deposited on  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> surface, suggesting that the electrons transfer from Au clusters to the  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> surface for Au/ $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> system, and the number of charge transfer between Au clusters and  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> for 4Au/ $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> is the largest (0.31e). The new peaks of complex refractive index, complex dielectric function and optical absorption due to Au clusters are observed in the visible range for Au-deposited  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> system, and the changing trend of their curves is similar. In addition, the intensities of complex refractive index, complex dielectric function and optical absorption are enhanced with increasing Au cluster size, showing a maximum value with 4Au. Increasing Au cluster size above 4Au results in a decrease in their intensities.

## Conflicts of interest

There are no conflicts of interest to declare.

## Acknowledgements

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## References

- 1 J. C. Little, A. T. Hodgson and A. J. Gadgil, *Atmos. Environ.*, 1993, **28**, 227–234.
- 2 W. Lindinger, A. Hansel and A. Jordan, *Int. J. Mass Spectrom. Ion Processes*, 1998, **173**, 191–241.
- 3 Y. Song, M. Shao, Y. Liu, S. Lu, W. Kuster, P. Goldan and S. Xie, *Environ. Sci. Technol.*, 2007, **41**, 4348–4353.
- 4 R. Atkinson and J. Arey, *Chem. Rev.*, 2003, **103**, 4605–4638.
- 5 T. Singh, N. Srivastava, P. K. Mishra, A. K. Bhatiya and N. L. Singh, *Mater. Sci. Forum*, 2016, **855**, 20–32.
- 6 K. Kante, M. Florent, A. Temirgaliyeva, B. Lesbayev and T. J. Badosz, *Carbon*, 2019, **146**, 568–571.
- 7 D. P. Das and K. M. Parida, *J. Mol. Catal. A: Chem.*, 2007, **276**, 17–23.
- 8 F. G. Edwards and N. Nirmalakhandan, *Water Sci. Technol.*, 1996, **34**, 565–571.
- 9 J. J. Pei and J. J. Zhang, *HVACR Res.*, 2011, **17**, 476–503.
- 10 M. Mishra, H. Park and D. M. Chun, *Adv. Powder Technol.*, 2016, **27**, 130–138.
- 11 G. Y. Zhang, Y. Feng, Y. Y. Xu, D. Z. Gao and Y. Q. Sun, *Mater. Res. Bull.*, 2012, **47**, 625–630.
- 12 M. Mishra and D. M. Chun, *Appl. Catal., A*, 2015, **498**, 126–141.
- 13 Y. Dong, L. Xing, F. Hu, A. Umar and X. Wu, *Vacuum*, 2018, **150**, 35–40.
- 14 F. J. Morin, *Phys. Rev.*, 1951, **83**, 1005–1010.
- 15 J. Kennedy and K. Frese, *J. Electrochem. Soc.*, 1978, **125**, 709–714.
- 16 H. B. Qi, D. W. Liu and H. Li, *Mater. Sci. Forum*, 2016, **852**, 14–21.
- 17 A. Quintanilla, N. Menéndez, J. Tornero, J. Casas and J. J. Rodríguez, *Appl. Catal., B*, 2008, **81**, 105–114.
- 18 H. Shan, C. Liu, L. Li, J. B. Zhang, H. Y. Li, Z. Liu, X. B. Zhang, X. Q. Bo and X. Chi, *ACS Appl. Mater. Interfaces*, 2013, **5**, 6376–6380.
- 19 B. Palanisamy, C. M. Babu, B. Sundaravel, S. Anandan and V. Murugesan, *J. Hazard. Mater.*, 2013, **252–253**, 233–242.
- 20 F. B. Li, X. Z. Li, C. S. Liu and T. X. Liu, *J. Hazard. Mater.*, 2007, **149**, 199–207.
- 21 G. K. Pradhan, S. Martha and K. M. Parida, *ACS Appl. Mater. Interfaces*, 2012, **4**, 707–713.
- 22 D. L. Allara, A. F. Hebard, F. J. Padden, R. G. Nuzzo and D. R. Falcone, *J. Vac. Sci. Technol., A*, 1983, **1**, 376–382.
- 23 J. Song, R. F. Guan, X. R. Liu, C. F. Jiang and G. H. Hou, *J. Phys. Chem. Solids*, 2018, **3697**, 31619–31620.
- 24 X. H. Liu, J. Zhang, X. Z. Guo, S. H. Wu and S. R. Wang, *Nanotechnology*, 2010, **21**, 095501.
- 25 D. D. Li, X. L. Yan, M. M. Yang, C. X. Luo, P. W. Li, J. Hu, G. Li, H. B. Jiang and W. D. Zhang, *J. Alloys Compd.*, 2018, **775**, 150–157.
- 26 H. X. Lin, Y. X. Liu, J. G. Deng, K. F. Zhang, X. Zhang, S. H. Xie, X. T. Zhao, J. Yang, Z. Han and H. X. Dai, *J. Environ. Sci.*, 2018, **70**, 74–86.
- 27 S. W. Cao, J. Fang, M. M. Shahjamali, Z. Wang and C. Xue, *CrystEngComm*, 2012, **14**, 7229–7235.
- 28 W. T. Chen, Y. J. Hsu and P. V. Kamat, *J. Phys. Chem. Lett.*, 2012, **3**, 2493–2499.
- 29 K. Jug, B. Zimmermann, P. Calaminici and A. M. Köster, *J. Chem. Phys.*, 2002, **116**, 4479.
- 30 M. Fang, X. L. Tan, Z. X. Liu, B. W. Hu and X. K. Wang, *Research*, 2021, **10**, 34133.
- 31 C. M. Aikens, S. Z. Li and G. C. Schatz, *J. Phys. Chem. C*, 2008, **112**, 11272–11279.
- 32 C. E. Zubieta, L. G. Aquino-Linare, S. A. Fuente, P. G. Beelli and R. M. Ferullo, *Comput. Mater. Sci.*, 2020, **173**, 109392.
- 33 S. W. Hon, L. Thomas, G. Jones and D. J. Willock, *Res. Chem. Intermed.*, 2015, **41**, 9587–9601.
- 34 M. T. Nguyen, M. F. Camellone and R. Gebauer, *J. Chem. Phys.*, 2015, **143**, 034704.
- 35 T. Pabisiak, M. J. Winiarski and A. Kiejna, *J. Chem. Phys.*, 2016, **144**, 044704.
- 36 M. C. Payne, M. P. Teter, D. C. Allan, T. A. Arias and J. D. Joannopoulos, *Rev. Mod. Phys.*, 1992, **64**, 1045–1097.
- 37 J. P. Perdew and W. Yue, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 1992, **45**, 13244–13249.
- 38 C. T. Chen, C. H. Zhao, X. Zhou, J. H. Chen, L. G. Chen and F. T. C. Li, *Vacuum*, 2021, **188**, 110164.
- 39 D. Vanderbilt, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 1990, **41**, 7892–7895.



- 40 M. C. Toroker, D. K. Kanan, N. Alidoust, L. Y. Isseroff, P. Liao and E. A. Carter, *Phys. Chem.*, 2011, **13**, 16644–16654.
- 41 M. Cao, T. Liu, S. Gao, G. Sun, X. Wu, C. Hu and Z. L. Wang, *Angew. Chem., Int. Ed.*, 2010, **36**, 4197–4201.
- 42 C. H. Zhao, B. S. Li, X. Zhou, J. H. Chen and H. Q. Tang, *Metals*, 2021, **11**, 424.
- 43 A. Kiejna and T. Pabisiak, *J. Phys.: Condens. Matter*, 2012, **24**, 095003.

