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# A highly sensitive surface-enhanced Raman scattering substrate prepared on a hydrophobic surface using controlled evaporation

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In the present work, we report the fabrication of a surface-enhanced Raman spectroscopy (SERS) substrate on a simple and easily fabricable hydrophobic surface. The substrates are prepared by slow and fast evaporation of a droplet of silver nanoparticle suspension in water. The corresponding identifiers for two substrates are "s\_evap" and "f\_evap" respectively. It is found that the dried spot size is small on s\_evap compared to that on f\_evap. This also minimizes the coffee stain effect and enriches the spot in a better way on s\_evap compared to f\_evap. Consequently, using SERS experimentation on our lab-built setup, concentration as low as  $2.5 \times 10^{-12}$  M of rhodamine 6G molecules was detected on s\_evap compared to  $2.5 \times 10^{-10}$  M on f\_evap. The proposed s\_evap SERS substrate is much easier to fabricate and easy to use compared to super-hydrophobic SERS substrates.

## 1. Introduction

Surface-enhanced Raman scattering (SERS) is a well-known variant of traditional Raman spectroscopy. Owing to its incredible sensitivity compared to conventional Raman spectroscopy, SERS has found extensive applications in diverse areas of science including bio-analysis,<sup>1,2</sup> environmental monitoring,<sup>3,4</sup> food science,<sup>5</sup> material characterization<sup>6</sup> and several other fields.<sup>7,8</sup> The enhancement of the Raman signal in SERS is attributed to electromagnetic and chemical enhancement. Among them, usually, electromagnetic enhancement is the dominating factor.<sup>9</sup> The source of the electromagnetic enhancement is the localized surface plasmon resonance (LSPR) which occurs from the excitation of electrons on the surface of metal nanostructures.<sup>10</sup> Since the field amplification on the surface of nanostructures depends on the type of metal (usually silver or gold), the size, and shape, the extent of the electromagnetic enhancement in SERS also depends on these parameters of nanostructures.<sup>11–16</sup> Several efforts have been made to fabricate SERS substrates using different sizes and shapes of Au and Ag nanoparticles for improved SERS efficiency. A very extensive discussion on these substrates can be found in recently published review articles.<sup>17,18</sup>

Apart from the metal nanostructure size and shape, an important condition for SERS to happen is the location of the target analyte molecules. The analyte molecules must be transported very close (within a few nm) to the nanostructures so that they can experience the enhanced electromagnetic field (usually called hot-spot) generated by nanoparticles.<sup>19</sup> A usual

process to prepare the substrate for SERS involves drop-casting of sample droplets containing colloidal nanoparticles and the target analyte molecule on a suitable surface. A drop casted on hydrophilic surfaces results in a coffee-ring effect with non-uniform distribution of analyte molecule as well as the nanoparticles. Usually, these are concentrated towards the periphery of the dried spot. The non-uniform distribution of nanoparticles and analyte molecules lowers the detection limit of the prepared substrates and the uniformity of the SERS signal.<sup>20,21</sup>

To overcome the coffee-ring effect and non-uniform distribution of nanoparticles and analyte molecules on the substrates, in recent years significant works have been reported. Most of these works concentrate on the fabrication of super-hydrophobic surfaces on which the contact area between droplet and substrate is very small compared to the size of the droplet resulting in small dried spot. This enables strong enrichment of the sample along with the suppression of the coffee-ring effect and, makes the detection of trace amount of analyte possible. The use of super-hydrophobic surfaces also reduces the quantity of samples to be used for substrate fabrication. By using the super-hydrophobic and nanoplasmonic structures, Angelis *et al.* have reported the detection of rhodamine 6G (Rh6G) up to attomolar ( $10^{-18}$  M) concentration.<sup>22</sup> However, the nanopillars used in this work require expensive equipment for its fabrication. In another approach, electrochemical deposition of metal nanostructures was performed to prepare super-hydrophobic substrates.<sup>23,24</sup> With the prepared substrate, the detection limit of fM ( $10^{-15}$  M) was achieved. Although the electrochemical method simplifies the process of fabrication of the nanostructures on the substrate surface, still the preparation process and involved equipment are relatively expensive. Very recently super-hydrophobic substrates for the

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SERS studies were developed from nature-inspired materials.<sup>25–27</sup> Although these super-hydrophobic surfaces show a strong enrichment and very low detection limit of samples, they can be used only for aqueous medium and requires excellent expertise for their fabrication.

Another strategy to overcome the coffee-ring effect is by using controlled evaporation<sup>28–31</sup> of a droplet on hydrophobic surfaces. As the controlled evaporation makes a uniform distribution of analyte, biosensing including SERS can be performed on these substrates.<sup>32</sup> Recently, Gerber *et al.* has shown that the Young modulus of the substrate surface and the relative humidity of the environment plays a critical role in the determination of evaporated drop diameter and so, the concentration of analyte in the dried drop.<sup>33</sup> Since the controlled evaporation method does not require either very good expertise as it is required in the fabrication of super-hydrophobic substrate or very expensive equipment, it can be easily utilized for the fabrication of SERS substrate.

In this work, we show that a simple hydrophobic surface with a controlled evaporation rate of a droplet can minimize the coffee-ring effect and enriches the analyte concentration. The substrate prepared with a slow evaporation rate under a high humidity environment is very much suitable for the surface-enhanced Raman scattering (SERS) of rhodamine 6G molecules. Compared to the super-hydrophobic surfaces, the making of a substrate on a hydrophobic surface is relatively simple and inexpensive and can be used for routine SERS studies.

## 2. Materials and methods

### 2.1 Materials

For the preparation of colloidal silver nanoparticles, silver nitrate (AgNO<sub>3</sub>), trisodium citrate dihydrate, sodium borohydride (NaBH<sub>4</sub>), sodium hydroxide (NaOH), and sodium chloride (NaCl) were procured from Sigma-Aldrich. For the fabrication of a hydrophobic surface, polydimethylsiloxane (PDMS: Sylgard 184) along with its curing agent was procured from Dow Corning, USA. All the chemicals were used as received.

### 2.2 Synthesis of silver nanoparticle

The spherical silver nanoparticle (size ~20 nm) were prepared by following the method reported by Agnihotri *et al.*<sup>34</sup> For the nanoparticle preparation, 48 mL of 1 mM NaBH<sub>4</sub> and 3.55 mM trisodium citrate solution in deionized water is heated at 60 °C for 30 min with continuous stirring. In this solution, 1 mL of 1 mM AgNO<sub>3</sub> is added dropwise using a syringe pump (200 μL min<sup>-1</sup>). The temperature of the solution is raised to 90 °C followed by the adjustment of pH to 10.5 using 100 mM NaOH solution. The reaction was continued till the change of color of the solution become evident. The prepared Ag nanoparticles were centrifuged and washed three times. The supernatant of the solution was removed and the nanoparticles were resuspended in deionized water. For the Ag nanoparticle aggregate formation, 200 μL of 2 M NaCl solution was mixed with the Ag nanoparticle solution and mixed gently. The solution was aged

for 24 hours prior to its use. The prepared Ag nanoparticle and aggregated nanoparticle were characterized using UV-vis spectroscopy with our lab-built setup.<sup>35</sup>

### 2.3 Fabrication of hydrophobic surface on glass

For the fabrication of a hydrophobic surface, PDMS solution with 50 : 1 ratio of pre-polymer and curing agent was prepared. The mixture is mixed thoroughly and degassed for 45 min to remove trapped air. The soda-lime glass slide (# 1.5, corning) was washed with soap followed by acetone–water solution. The cleaned glass surface is coated with PDMS using spin coating at 3000 rpm for 2 min. The coated glass slide was cured at 85 °C for 45 min in oven. For the characterization of the prepared PDMS surface, contact angle measurements were performed. For all the contact angle measurements, a water drop of ~4 μL was used. All the contact angle measurements were carried out with our lab-built setup.<sup>36</sup> The analysis of contact angle was performed using the ‘drop analysis’ plugin in ImageJ.<sup>37,38</sup>

### 2.4 SERS substrate preparation

The surface-enhanced Raman scattering substrate was prepared using the aggregated silver nanoparticles in water. The nanoparticle aggregates were drop casted on a cleaned hydrophilic glass surface and PDMS coated hydrophobic glass surface. On the PDMS coated surface, drop evaporation rate was manipulated by controlling the relative humidity of the environment. A high evaporation rate of the casted drop was obtained at relative humidity ~50% whereas slow evaporation of the drop was achieved at >85% relative humidity. The slow evaporation of drop was performed in a humid chamber made using an airtight desiccator. The sample drop on the hydrophilic glass surface was evaporated under an ambient environment. In the following sections, the substrate prepared using slow and fast evaporation rates are named as “s\_evp” and “f\_evp” respectively.

### 2.5 Raman spectroscopy instrumentation and SERS experiments

All the Raman and SERS experiments reported in this work were performed using our inexpensive lab-built Raman spectroscopy setup. The schematic of the developed Raman setup is shown in Fig. 1. The setup used is a modified version of our earlier reported setup.<sup>39,40</sup> Briefly, the Raman setup utilized a 638 nm laser diode (LD) controlled with a variable current DC power source (DCP). The power of the laser diode can be controlled either by varying the operating current or by applying a polarizer (pol) in the beam path. The light beam is focused (using FL1) on a silver-coated mirror with a hole (M1, hole dia: 3 mm) in its center. The mirror M1 also serves the purpose of collection of back-scattered Raman signal. The light beam is made to focus on the sample (S) mounted on xyz stage using a silver-coated mirror M2 and a microscope objective (mag.: 20×). The back-scattered Raman signal was diverted from M1 towards the long-pass filter (LPF, Omega-opticals) and collecting lens (CL). The collected signal is then fed to TE cooled spectrometer (Avantes: AvaSpec-ULS2048LTEC, wavelength range 520–1000 nm,



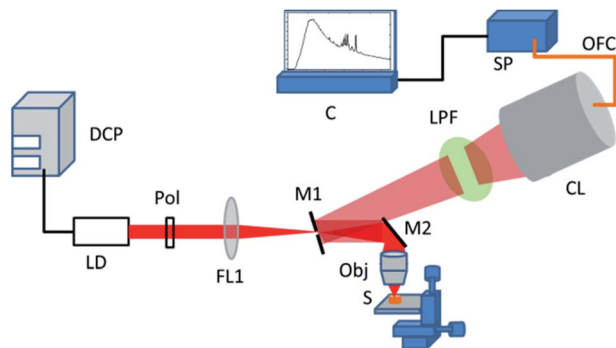


Fig. 1 Schematic of the lab-built Raman spectroscopy setup. The abbreviations of components are as DCP: DC power supply, LD: laser diode, Pol: polarizer, FL1: focusing lens, M1: mirror with hole, M2: mirror, Obj: microscope objective, LPF: long pass filter, CL: collecting lens, OFC: optical fiber cable, SP: spectrometer, C: computer.

resolution: 0.6 nm) using a fiber-optic patch cord. The recording of Raman spectra was performed using Avasoft software interfaced with the Avantes spectrometer.

For Raman and SERS experiments, rhodamine 6G (Rh6G) in water was used as a probe molecule. For recording of SERS spectra, 3  $\mu\text{L}$  of colloidal silver nanoparticle aggregate was evaporated on the prepared glass and PDMS surfaces. On the dried nanoparticle spot, 3  $\mu\text{L}$  of various concentrations of the Rh6G was drop casted and evaporated. For the recording of Raman spectrum, a 3  $\mu\text{L}$  of 100 mM Rh6G solution in water was drop casted and evaporated on the PDMS coated glass coverslip. For measurement of SERS spectra, uniform experimental parameters were maintained as LD operating current: 120 mA, voltage 6.0 V, acquisition time: 3 s, and onboard averaging: 3. For Raman spectroscopy of Rh6G, an acquisition time of 10 s with 3 onboard averaging was used. The final SERS spectra were obtained by performing baseline correction.

## 3. Results and discussion

### 3.1 Characterization of hydrophobic surface and sample evaporation

The PDMS surface prepared using pre-polymer and curing agent (50 : 1 ratio) is characterized using contact angle measurement. For the contact angle measurement, 4  $\mu\text{L}$  of deionized water droplet is used. Fig. 2(a) (i) shows the droplet topology on the prepared PDMS surface. The water contact angle is found to be  $\sim 94$  degrees. The series of images shown in Fig. 2(a) shows the evaporation of water droplet with time on the PDMS surface at room temperature and relative humidity 50%. The initial contact diameter (in pixel) for the drop was found to be 95 pixels. Progressive evaporation shows changes in the contact angle as well as in the drop contact diameter. The contact angle reaches 12–13 degrees before the drop completely disappears. The contact diameter measured just before the complete evaporation of the drop was found to be 65 pixels. Fig. 2(b) shows the evaporation of water drop with time under the high humidity environment. The images in the figure show that although the drop contact diameter decreases during the

evaporation process, the contact angle does not show appreciable change. Further, as expected complete evaporation of drop takes relatively longer time. The drop contact diameter on the surface after 100 min of evaporation was found to be  $\sim 40$  pixels. Fig. 2(c) and (d) shows the microscope images of dried drop obtained with fast and slow evaporation respectively. As it can be seen, the central part of the dried spot shown in Fig. 2(d) shows higher concentration of deposited nanoparticles. Fig. 2(e) shows the change in the drop contact diameter with evaporation time. It is evident from the plot that the drop diameter decreases to lower value in high humid environment compared to moderate humid environment.

It has been observed for these kinds of soft PDMS surfaces, under low or medium relative humidity, the water contact line is pinned to the surface and therefore, the contact diameter stays nearly constant while the contact angle decreases rapidly.<sup>33</sup> The observed behavior is in contrast to the usual rigid PDMS surface where the ratio of pre-polymer and curing agent is 9 : 1. A higher ratio of pre-polymer and curing agent lowers the Young's modulus very strongly (20 kPa) compared to rigid PDMS (2937 kPa) surface.<sup>33</sup> The observed steady decrease in the contact angle under 50% relative humidity results from the lowering of Young's modulus.<sup>33</sup> It was also observed that the magnitude of stress vector field  $|\sigma(n)|$  increases during the evaporation process for both fast and slow evaporation. However, the value of it exceeds for fast evaporation compared to its value during slow evaporation. This leads to accumulation of substrate stress leading to change in the wetting ridge shape and consequently the contact angle compared to the slow evaporation process. Overall, this results in a smaller dried drop diameter at high relative humidity with a higher concentration of the nanoparticle solution (in water) and analyte molecules. This could be the basis of better SERS signal obtained with substrate prepared on hydrophobic surfaces using slow evaporation rate.

### 3.2 Surface enhanced Raman spectroscopy on hydrophobic surfaces

For the SERS measurements, aggregated spherical Ag nanoparticles of size  $\sim 20$  nm were used in the present work. An earlier reported work on SERS activity of spherical Ag nanoparticle indicates that the optimum SERS signal can be obtained with particle size  $> 40$  nm.<sup>15</sup> It is also well known that the aggregation of Ag nanoparticle enhances the SERS signal strength.<sup>41</sup> The aggregation of nanoparticles not only induces hot-spot but also increases the effective size of the nanoparticle. The aggregation of 20 nm nanoparticle in this work can lead to larger effective size ( $> 40$  nm) of the cluster along with the availability of hot-spots. The SERS substrates were prepared using these Ag nanoparticle aggregates on the PDMS coated glass coverslips. The Ag nanoparticles (size  $\sim 20$  nm) were prepared using the method proposed by Agnihotri *et al.*<sup>34</sup> The extinction spectrum of prepared nanoparticle is shown in Fig. 3. As it is evident, the spectrum shows a single LSPR band characteristic of spherical Ag nanoparticles. To enhance the SERS activity, nanoparticles were aggregated by adding 200  $\mu\text{L}$  of 2 M NaCl in water. The extinction spectrum of nanoparticles after



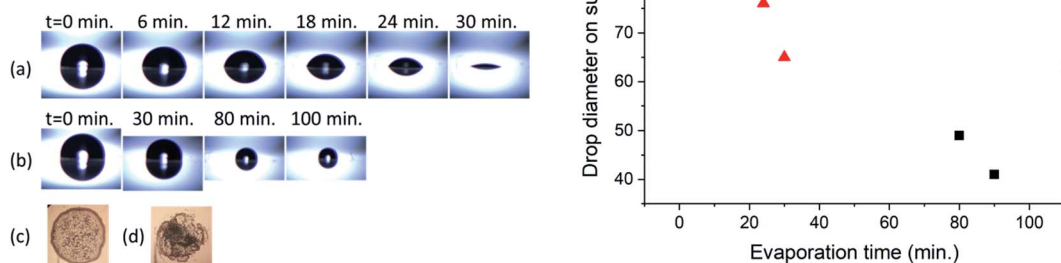


Fig. 2 (a) Fast droplet evaporation on prepared PDMS surface (b) slow droplet evaporation on prepared PDMS surface (c) dried spot for evaporation in ambient environment (d) dried spot for slow evaporation (e) plot of drop contact diameter against time for slow and fast evaporation processes.

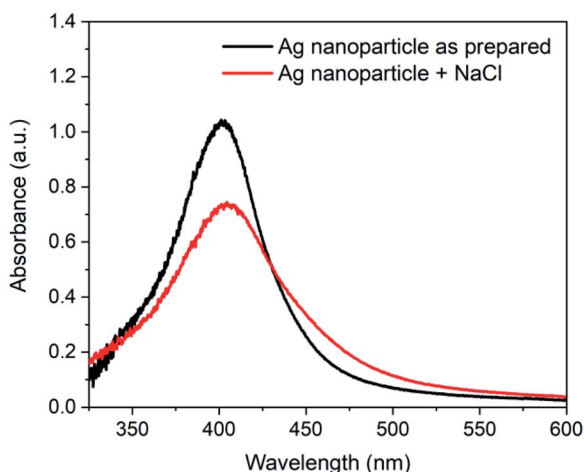


Fig. 3 UV-vis spectrum of Ag nanoparticle as prepared and after addition of NaCl.

aggregation is also shown in Fig. 3. As it can be seen, the aggregation of Ag nanoparticle results in the broadening of the LSPR band with decreased absorbance.

For SERS studies, rhodamine 6G (Rh6G) was used as a probe molecule to show the SERS enhancement on the substrate  $s_{\text{evp}}$ . This molecule has been extensively used as a model molecule in several SERS studies. The Rh6G easily get adsorbed on the Ag nanoparticle surface and induces aggregation<sup>42,43</sup> which may further improve the SERS signal. To show the enhancement of the SERS signal, spectra of Rh6G were recorded on a cleaned glass coverslip,  $f_{\text{evp}}$ , and  $s_{\text{evp}}$  substrates. Fig. 4 shows the comparison of corresponding SERS spectra. In the bottom panel, Raman spectrum of Rh6G (conc. 100 mM) is also shown. In all three SERS spectra, Rh6G concentration of 2.5  $\mu\text{M}$  was utilized. The intensity difference in the SERS spectra is clearly evident from the figure. In the spectra, maximum intensity of bands can be seen for  $s_{\text{evp}}$  substrate followed by the  $f_{\text{evp}}$  and sample on the hydrophilic glass surface. In all the spectra, characteristic bands of rhodamine 6G *e.g.*, 616 (C–C

in-plane bend in xanthene ring), 776 (out-of-plane C–H bend), 1181 (C–H bend in xanthene ring), 1311 (xanthene ring breathing), 1361 (C–C stretch), 1509 (C–C stretch in xanthene ring, C–N stretch, in-plane C–H bend, in-plane N–H bend), 1569 (C–C stretch in xanthene ring, in-plane N–H bend), and 1648  $\text{cm}^{-1}$  (C–C stretch in xanthene ring, in-plane C–H bend) can be easily seen.<sup>44,45</sup> In the case of substrate made on a cleaned glass slide, the maximum Raman signal was obtained at the periphery whereas optimum Raman signal for  $f_{\text{evp}}$  and  $s_{\text{evp}}$  were obtained between the periphery and the center. Although the evaporated sample spot is of smaller dimension in the slow evaporation process, it appears that the coffee ring

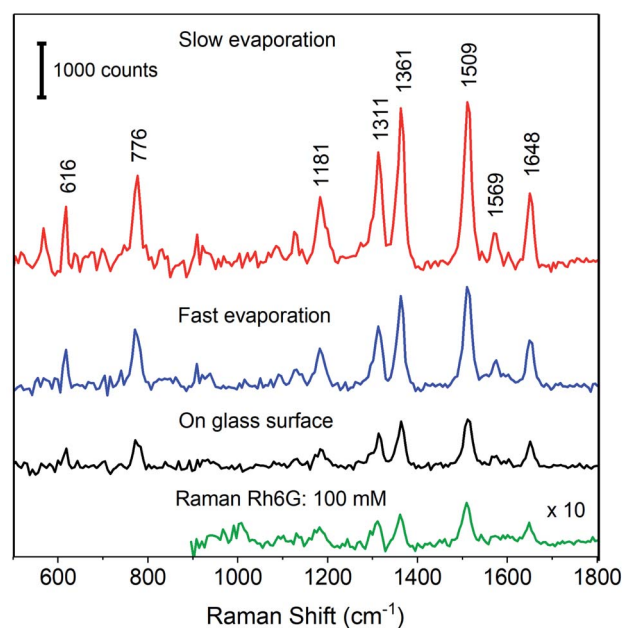


Fig. 4 SERS spectra of Rh6G on surfaces prepared using slow evaporation ( $s_{\text{evp}}$ ), fast evaporation ( $f_{\text{evp}}$ ) and on hydrophilic cleaned glass surface. The concentration of Rh6G in all cases is 2.5  $\mu\text{M}$ . Bottom panel: Raman spectrum of Rh6G (conc. 100 mM).



effect is not completely suppressed. Due to this, the SERS signal at the center of the dried spot was relatively lower compared to the region between the periphery and the center. To evaluate the performance of substrate prepared under slow evaporation, SERS signal strength must be compared with the SERS signal obtained with substrate prepared under fast evaporation and, also with the sample on hydrophilic glass surface. To achieve this, although the SERS signal was observed at various locations, the spectra reported here are obtained from single point after optimization of the SERS signal strength. Compared to the SERS signal on a glass substrate, the SERS signal is enhanced  $\sim 2$  times on *f\_evp* substrate and  $\sim 3.5$  times on *s\_evp* substrate.

In order to investigate the detection sensitivity on PDMS surfaces, SERS spectra were recorded with various concentrations of Rh6G. For experiments, seven sample spots were prepared with Rh6G concentration ranging from 2.5  $\mu\text{M}$  to 2.5 pM. Fig. 5(a) shows the SERS spectra of Rh6G with concentrations ranging from 2.5  $\mu\text{M}$  to 25 pM on *f\_evp* substrate. Fig. 5(b) shows the SERS spectra for the Rh6G from 2.5  $\mu\text{M}$  to 2.5 pM from the spot obtained on the *s\_evp* substrate. It can be clearly seen from Fig. 5(b) that SERS signals were observed from the enriched regions in the drop-casted sample spots and, even with 2.5 pM concentration, the characteristic Raman bands of Rh6G were clearly observed. Observation of SERS signal from each spot confirms the usability and reproducibility of the sample preparation method using slow evaporation. As it is mentioned above, for comparison of SERS strengths on substrates *f\_evp* and *s\_evp*, single point measurement was performed with maximization of the SERS signal. As it can be seen, the lower concentration with clearly observable Raman bands in the case of *f\_evp* substrate was  $\sim 250$  pM. For 25 pM concentration, a very small band appears at  $\sim 1609$   $\text{cm}^{-1}$  with no trace of other characteristic bands. The intensity of this band is comparable to noise in the spectrum. Therefore, this spectrum was not considered for further analysis. This observed change indicates towards better enrichment of the substrate

surface using a slow evaporation process, compared to that obtained with fast evaporation process. In the bottom panel of Fig. 5(a), the Raman spectrum of Rh6G is shown. The spectrum was obtained from dried drop of concentration 100 mM with acquisition time and onboard averaging of 10 and 3 respectively. The Raman spectrum showed in the figure is expanded 10-fold on intensity scale to make the clear appearance of bands.

Fig. 6(a) and (b) shows the variation in the peak intensity of the band observed at 1509  $\text{cm}^{-1}$ , on *f\_evp* and *s\_evp* substrates respectively. As it can be seen, in Fig. 6(a), the intensity of the band decreases linearly whereas the decrease is linear only up to 0.25 nM concentration in Fig. 6(b). Below the concentration 0.25 nM in Fig. 6(b), the intensity decrease is slow. The linear fit to the intensity is also shown in the figure. The obtained linear correlation for *f\_evp* substrate was  $Y = 355 \times \text{Log } C_{\text{Rh6G}} + 3637.5$  whereas for *s\_evp* substrate, it is  $Y = 479 \times \text{Log } C_{\text{Rh6G}} + 5119$ . The results show that the enrichment of the sample is better when it is slowly evaporated, and detection of Rh6G molecule up to 2.5 pM can be observed clearly.

The enhancement factor was calculated based on the SERS spectra obtained at 2.5  $\mu\text{M}$  concentration using,

$$\text{AEF} = \frac{I_{\text{SERS}}/C_{\text{SERS}}}{I_{\text{Raman}}/C_{\text{Raman}}}$$

where the  $I_{\text{SERS}}$  and  $C_{\text{SERS}}$  are the intensity of band at 1509  $\text{cm}^{-1}$  and concentration of analyte molecule respectively. The  $I_{\text{Raman}}$  and  $C_{\text{Raman}}$  are intensity of same band in the Raman spectrum and the concentration of analyte molecule. For the Raman spectrum, Rh6G in concentration of 100 mM was used. The recorded Raman spectrum was with higher acquisition time compared to the SERS spectra. To compensate the difference in the integration time, the intensity was normalized to counts per sec. The calculated enhancement factor for the *f\_evp* substrate was  $2 \times 10^6$  whereas, for the *s\_evp* substrate, it was  $3.9 \times 10^6$ . Therefore, the enhancement factor increases by nearly

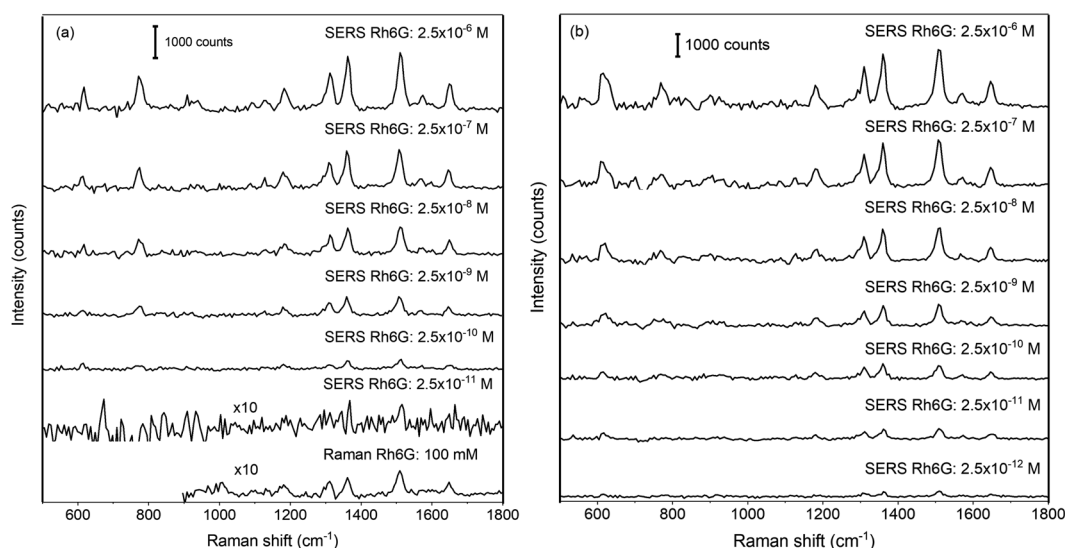


Fig. 5 Surface enhanced Raman spectra of rhodamine 6G on hydrophobic (a) *f\_evp* and (b) *s\_evp* substrates.



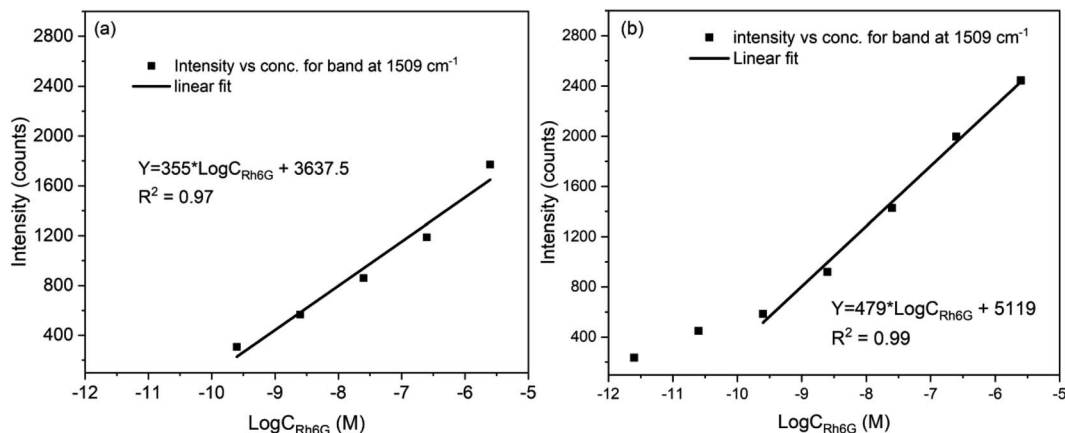


Fig. 6 Variation of intensity of band at 1509 cm<sup>-1</sup> with concentration of Rh6G on substrate obtained by (a) fast evaporation and (b) slow evaporation.

100% on the s<sub>evp</sub> substrate. This very clearly emphasizes the improvement in the SERS signal with an easily fabricated hydrophobic PDMS coated glass surface. The controlled evaporation of sample on the hydrophobic surface is useful for the enrichment of the sample and hence for the enhancement of the SERS signal.

## 4. Conclusions

A soft hydrophobic surface using PDMS was prepared using pre-polymer and curing agent ratio of 50 : 1. It is shown that the evaporation rate on this PDMS surface controls the dried drop size and the uniformity of analyte distribution in the spot. The slow evaporation of nanoparticle suspension in water under high humid environment minimizes the coffee stain effect and enriches the dried drop compared to the fast evaporation process under low humidity. The substrate prepared with slow evaporation is found to be suitable for the surface enhanced Raman spectroscopy of Rh6G, and concentration as low as 2.5 × 10<sup>-12</sup> M was detected. The enhancement factor for the SERS measurement was found to be two time compared to the substrate prepared under low humidity condition. The successful demonstration of the use of simple and easily fabricable hydrophobic surface for the SERS studies is promising for further applications in sensitive and reliable detection of analyte molecules with SERS technique. The sensitivity of the SERS measurement on used soft hydrophobic surface can be further enhanced by using anisotropic Ag nanoparticles with optimized size and shape. This will paves the way for the sensitive detection of analyte molecules other than Rh6G depending on their SERS cross-section.

## Conflicts of interest

There are no conflicts to declare.

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## References

- 1 C. Zong, M. Xu, L.-J. Xu, T. Wei, X. Ma, X.-S. Zheng, R. Hu and B. Ren, *Chem. Rev.*, 2018, **118**, 4946–4980.
- 2 H. Chen, A. Das, L. Bi, N. Choi, J.-I. Moon, Y. Wu, S. Park and J. Choo, *Nanoscale*, 2020, **12**, 21560–21570.
- 3 D. Song, R. Yang, F. Long and A. Zhu, *J. Environ. Sci.*, 2019, **80**, 14–34.
- 4 H. Wei, S. M. Hossein Abtahi and P. J. Vikesland, *Environ. Sci.: Nano*, 2015, **2**, 120–135.
- 5 R. Pilot, *J. Raman Spectrosc.*, 2018, **49**, 954–981.
- 6 S. Dick, M. P. Konrad, W. W. Y. Lee, H. McCabe, J. N. McCracken, T. M. D. Rahman, A. Stewart, Y. Xu and S. E. J. Bell, *Adv. Mater.*, 2016, **28**, 5705–5711.
- 7 J. Wu, L. Zhang, F. Huang, X. Ji, H. Dai and W. Wu, *J. Hazard. Mater.*, 2020, **387**, 121714.
- 8 I. J. Jahn, A. Mühlhig and D. Cialla-May, *Anal. Bioanal. Chem.*, 2020, **412**, 5999–6007.
- 9 J. R. Lombardi and R. L. Birke, *Acc. Chem. Res.*, 2009, **42**, 734–742.
- 10 S.-Y. Ding, E.-M. You, Z.-Q. Tian and M. Moskovits, *Chem. Soc. Rev.*, 2017, **46**, 4042–4076.
- 11 M. F. Cardinal, E. Vander Ende, R. A. Hackler, M. O. McAnally, P. C. Stair, G. C. Schatz and R. P. Van Duyne, *Chem. Soc. Rev.*, 2017, **46**, 3886–3903.
- 12 G. Demirel, H. Usta, M. Yilmaz, M. Celik, H. A. Alidagi and F. Buyukserin, *J. Mater. Chem. C*, 2018, **6**, 5314–5335.
- 13 H. K. Lee, Y. H. Lee, C. S. L. Koh, G. C. Phan-Quang, X. Han, C. L. Lay, H. Y. F. Sim, Y.-C. Kao, Q. An and X. Y. Ling, *Chem. Soc. Rev.*, 2019, **48**, 731–756.
- 14 F. Tian, F. Bonnier, A. Casey, A. E. Shanahan and H. J. Byrne, *Anal. Methods*, 2014, **6**, 9116–9123.



- 15 K. G. Stamplecoskie, J. C. Scaiano, V. S. Tiwari and H. Anis, *J. Phys. Chem. C*, 2011, **115**, 1403–1409.
- 16 Y. Yang, S. Matsubara, L. Xiong, T. Hayakawa and M. Nogami, *J. Phys. Chem. C*, 2007, **111**, 9095–9104.
- 17 J. Langer, D. Jimenez de Aberasturi, J. Aizpurua, R. A. Alvarez-Puebla, B. Auguié, J. J. Baumberg, G. C. Bazan, S. E. J. Bell, A. Boisen, A. G. Brolo, J. Choo, D. Cialla-May, V. Deckert, L. Fabris, K. Faulds, F. J. García de Abajo, R. Goodacre, D. Graham, A. J. Haes, C. L. Haynes, C. Huck, T. Itoh, M. Käll, J. Kneipp, N. A. Kotov, H. Kuang, E. C. Le Ru, H. K. Lee, J.-F. Li, X. Y. Ling, S. A. Maier, T. Mayerhöfer, M. Moskovits, K. Murakoshi, J.-M. Nam, S. Nie, Y. Ozaki, I. Pastoriza-Santos, J. Perez-Juste, J. Popp, A. Pucci, S. Reich, B. Ren, G. C. Schatz, T. Shegai, S. Schlücker, L.-L. Tay, K. G. Thomas, Z.-Q. Tian, R. P. Van Duyne, T. Vo-Dinh, Y. Wang, K. A. Willets, C. Xu, H. Xu, Y. Xu, Y. S. Yamamoto, B. Zhao and L. M. Liz-Marzán, *ACS Nano*, 2020, **14**, 28–117.
- 18 C. Li, Y. Huang, X. Li, Y. Zhang, Q. Chen, Z. Ye, Z. Alqarni, S. E. J. Bell and Y. Xu, *J. Mater. Chem. C*, 2021, **9**, 11517–11552.
- 19 H. Liu, L. Yang and J. Liu, *TrAC, Trends Anal. Chem.*, 2016, **80**, 364–372.
- 20 X. Pan, J. Dong, Y. Li, X. Sun, C. Yuan and W. Qian, *RSC Adv.*, 2016, **6**, 29586–29591.
- 21 N. Pazos-Perez, C. S. Wagner, J. M. Romo-Herrera, L. M. Liz-Marzán, F. J. García de Abajo, A. Wittmann, A. Fery and R. A. Alvarez-Puebla, *Angew. Chem., Int. Ed.*, 2012, **51**, 12688–12693.
- 22 F. De Angelis, F. Gentile, F. Mecarini, G. Das, M. Moretti, P. Candeloro, M. L. Coluccio, G. Cojoc, A. Accardo, C. Liberale, R. P. Zaccaria, G. Perozziello, L. Tirinato, A. Toma, G. Cuda, R. Cingolani and E. Di Fabrizio, *Nat. Photonics*, 2011, **5**, 682–687.
- 23 Y. Gao, T. You, N. Yang, C. Zhang and P. Yin, *Adv. Mater. Interfaces*, 2019, **6**, 1801966.
- 24 H. Li, Q. Yang, J. Hou, Y. Li, M. Li and Y. Song, *Adv. Funct. Mater.*, 2018, **28**, 1800448.
- 25 B.-B. Xu, Y.-L. Zhang, W.-Y. Zhang, X.-Q. Liu, J.-N. Wang, X.-L. Zhang, D.-D. Zhang, H.-B. Jiang, R. Zhang and H.-B. Sun, *Adv. Opt. Mater.*, 2013, **1**, 56–60.
- 26 P. Kumar, R. Khosla, M. Soni, D. Deva and S. K. Sharma, *Sens. Actuators, B*, 2017, **246**, 477–486.
- 27 G. C. Shi, M. L. Wang, Y. Y. Zhu, L. Shen, W. L. Ma, Y. H. Wang and R. F. Li, *Sci. Rep.*, 2018, **8**, 6916.
- 28 Y. Li, C. Diddens, T. Segers, H. Wijshoff, M. Versluis and D. Lohse, *Proc. Natl. Acad. Sci. U. S. A.*, 2020, **117**, 16756.
- 29 A. Gao, J. Liu, L. Ye, C. Schönecker, M. Kappl, H.-J. Butt and W. Steffen, *Langmuir*, 2019, **35**, 14042–14048.
- 30 M. Majumder, C. S. Rendall, J. A. Eukel, J. Y. L. Wang, N. Behabtu, C. L. Pint, T.-Y. Liu, A. W. Orbaek, F. Mirri, J. Nam, A. R. Barron, R. H. Hauge, H. K. Schmidt and M. Pasquali, *J. Phys. Chem. B*, 2012, **116**, 6536–6542.
- 31 C. Liu, E. Bonaccorso and H.-J. Butt, *Phys. Chem. Chem. Phys.*, 2008, **10**, 7150–7157.
- 32 Y. Wang, F. Liu, Y. Yang and L.-P. Xu, *Mater. Chem. Front.*, 2021, **5**, 5639–5652.
- 33 J. Gerber, T. Lendenmann, H. Eghlidi, T. M. Schutzius and D. Poulidakos, *Nat. Commun.*, 2019, **10**, 4776.
- 34 S. Agnihotri, S. Mukherji and S. Mukherji, *RSC Adv.*, 2014, **4**, 3974–3983.
- 35 H. Hegde, C. Santhosh and R. K. Sinha, *Mater. Res. Express*, 2019, **6**, 105075.
- 36 R. K. Sinha, *Laser Phys.*, 2019, **30**, 026202.
- 37 C. A. Schneider, W. S. Rasband and K. W. Eliceiri, *Nat. Methods*, 2012, **9**, 671–675.
- 38 A. F. Stalder, T. Melchior, M. Müller, D. Sage, T. Blu and M. Unser, *Colloids Surf., A*, 2010, **364**, 72–81.
- 39 R. K. Sinha, *Instrum. Exp. Tech.*, 2021, **64**, 840–847.
- 40 R. K. Sinha and P. Biswas, *J. Mol. Struct.*, 2020, **1222**, 128946.
- 41 R. Pilot and M. Massari, *Chem. Phys.*, 2021, **2**, 100014.
- 42 W. E. Smith and C. Rodger, in *Encyclopedia of Spectroscopy and Spectrometry*, ed. J. C. Lindon, Academic Press, Oxford, 2nd edn, 1999, pp. 2822–2827, DOI: 10.1016/B978-0-12-374413-5.00304-3.
- 43 S. Kruszewski and M. Cyrankiewicz, *Acta Phys. Pol., A*, 2013, **123**, 965–969.
- 44 P. Hildebrandt and M. Stockburger, *J. Phys. Chem.*, 1984, **88**, 5935–5944.
- 45 L. Jensen and G. C. Schatz, *J. Phys. Chem. A*, 2006, **110**, 5973–5977.

