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The enhanced photocatalytic performance and first-principles computational insights of Ba doping-dependent TiO₂ quantum dots

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Degradation in the presence of visible light is essential for successfully removing dyes from industrial wastewater, which is pivotal for environmental and ecological safety. In recent years, photocatalysis has emerged as a prominent technology for wastewater treatment. This study aimed to improve the photocatalytic efficiency of synthesized TiO₂ quantum dots (QDs) under visible light by barium (Ba) doping. For this, different weight ratios (2% and 4%) of Ba-doped TiO₂ QDs were synthesized under ambient conditions via a simple and modified chemical co-precipitation approach. The QD crystal structure, functional groups, optical features, charge-carrier recombination, morphological properties, interlayer spacing, and presence of dopants were analyzed. The results showed that for 4% Ba-doped TiO₂, the effective photocatalytic activity in the degradation process of methylene blue (MB) dye was 99.5% in an alkaline medium. Density functional theory analysis further corroborated that the band gap energy was reduced when Ba was doped into the TiO₂ lattice, implying a considerable redshift of the absorption edge due to in-gap states near the valence band.

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1. Introduction

Freshwater reservoirs are getting more drained and polluted due to their use in various industrial fields, notably garment industries. Freshwater supplies are constantly getting depleted due to population growth and extended droughts, and the world now faces a massive problem in fulfilling the expanding needs for potable water.^{1–4} Approximately 750 million people on the planet are suffering from a shortage of clean water, and an increase in the supply of micropollutants in water has proven dangerous to the environment. Industries

including textiles, polycarbonate, paper printing, foodstuff, and chemicals are the primary water pollution sources.^{5–7} Industrial dyes can cause major sicknesses in animals and humans, such as cancer, skin irritability, numerous respiratory ailments, allergies, and liver malfunctions.⁸ It is reported that 15% of dyes are directly discharged into the environment. Some traditional approaches, including advanced oxidation processes,⁹ adsorption,¹⁰ flocculation and coagulation,¹¹ electrocoagulation,¹² and photocatalytic degradation,¹³ are utilized in wastewater treatment. These methods have some flaws, such as inadequate removal and the formation of secondary contaminants. Besides these, photocatalytic activity (PCA) in the presence of metal oxide semiconductors is recognized as an effective approach for removing contaminants from wastewater.¹⁴

Many metal oxide semiconductors (Fe₂O₃, TiO₂, MgO, and WO₃) are widely used for photocatalytic dye degradation due to their excellent chemical stabilization, low toxicity, and broad band gap. Among them, titanium dioxide (TiO₂) has versatile characteristics such as a large band gap, high physiochemical stability, zero toxicity, high oxidative power, inexpensiveness, effective bactericidal potential, and photocatalytic activity under UV light.^{15–17} TiO₂ naturally exists in three crystalline phases: anatase (tetragonal), rutile (tetragonal), and brookite (orthorhombic).^{18,19} Among these, strong photocatalytic activity, chemical stability, and a negative charge-carrier potential are credited with making the anatase phase one of the most

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effective photocatalysts.²⁰ The physicochemical parameters, surface-to-volume ratio, grain size, geometry, crystallinity, and surface texture influence the photocatalytic activity.^{21,22}

The development of new mixed metal oxides (MOs) for degrading various organic dyes by photocatalysis has been a major focus in recent years. As a result, doping TiO₂ with noble metals, rare-earth metals, and transition metals has been extensively studied to reduce the particle size and improve the photocatalytic and antibacterial properties of TiO₂. Doping TiO₂ with the alkaline earth metal Ba alters the oxide's optical absorption characteristics, which may affect its antibacterial activity and PCA. By modifying the band gap energy, the Ba dopant may operate as an electron trap, inhibiting the recombination rate of e⁻/h⁺ pairs. This enhances the antibacterial action and PCA by promoting the generation of reactive oxygen species.²³ According to Mills and Hoffmann's research, the photocatalytic efficiency of a nanocatalyst can be improved by doping the nanocatalyst with an appropriate substance.²⁴ However, few studies are reported in the literature regarding Ba-doped TiO₂ synthesized through different methodologies. In a recent study, up to 3% Ba-doped TiO₂ was synthesized using the sol-gel method at 80 °C for 5 hours. The product was calcined at 500 °C for 2 hours and a nanoparticle morphology was obtained. The acquired nanoparticles were utilized for the degradation of methylene blue (MB) dye for up to 150 minutes, and the calculated degradation percentage was insufficient.²⁵ In another study, up to 2% barium-doped TiO₂ was prepared using a mild hydrothermal method and placed in an oven at 100 °C for 12 hours to complete the chemical reaction. After drying the product at 50 °C, the nanocrystal morphology was determined and used for the degradation of acid red 18, a dye. The degradation efficiency was calculated to be 98.6% under acidic conditions for 90 minutes.²⁶ In this regard, pure TiO₂ and Ba-

doped TiO₂ in the anatase form are particularly interesting among the various photoactive MOs.

Herein, 2% and 4% Ba-doped TiO₂ were prepared with a simple co-precipitation technique and a QD morphology was achieved. We determined the maximum doping effect of Ba on TiO₂ using structural, optical, and morphological techniques and tested its ability to remove organic pollutants in different media from contaminated water. The obtained results showed maximum degradation in almost all media; however, the most promising results were in the basic medium and were found to be 99.5%. Furthermore, computational density functional theory was employed to investigate the Ba-doped TiO₂ systems. We show that the changes in the electronic structure of TiO₂ doped with Ba can be explained by combining experimental results with theoretical calculations.

2. Experimental section

2.1 Materials

Titanium(IV) butoxide (C₁₆H₃₆O₄Ti), ethanol (C₂H₅OH), sodium hydroxide (NaOH), and barium chloride dehydrate (BaCl₂·2H₂O) were purchased from Sigma-Aldrich (Germany).

2.2 Synthesis of Ba-doped TiO₂ quantum dots

To synthesize TiO₂, titanium(IV) butoxide (20 mL) was dissolved in ethanol (50 mL) under vigorous stirring at 75 °C to form a solution. Later, 400 mL deionized water (DI water) and NaOH (0.5 M) were introduced dropwise to form a colloidal solution; the formation of precipitates started during the incorporation of the 0.5 M NaOH solution while maintaining the above-mentioned colloidal solution at pH ~ 10. Then, the colloidal solution was centrifuged for 6 minutes at 7500 rpm to achieve sedimentation, and the product was rinsed numerous times

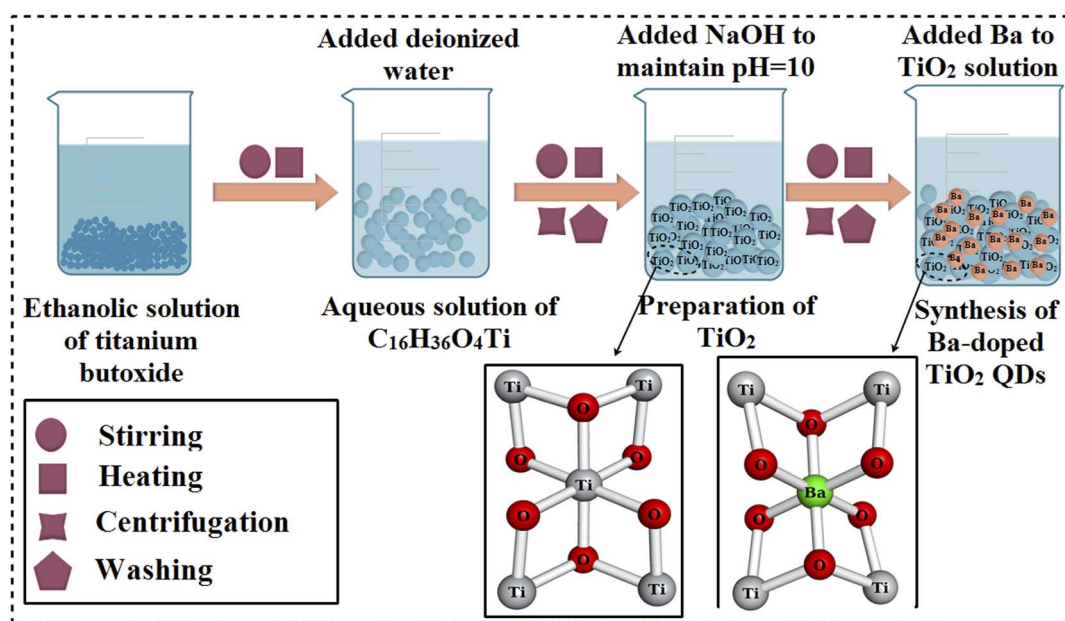


Fig. 1 Schematic diagram of the synthesized Ba-doped TiO₂ QDs.



with deionized (DI) water to eliminate impurities. The final product was dried at 500 °C to create anatase-phase TiO₂ fine powder using the co-precipitation technique. For the doping of Ba into TiO₂, different concentrations of Ba (2 and 4 wt%) were incorporated in the colloidal solution of TiO₂ QDs, and the same procedure was adapted for the synthesis of doped QDs (Fig. 1).

2.3 Photocatalysis

The PCA of the TiO₂ and Ba-doped TiO₂ QDs under the influence of light ranging between 400 and 800 nm at room temperature was evaluated with UV-vis spectrophotometry. Initially, 5 mg of the nanocatalyst was dissolved in a 30 mL aqueous solution of MB while being vigorously stirred for 10 minutes in the absence of light. Then, the solution was exposed to the above-mentioned light for 2 hours, with 3 mL of the solution taken out every 30 minutes for UV-vis analysis to evaluate the MB dye degradation ($\lambda_{\text{max}} = 665 \text{ nm}$). The percentage degradation efficiency was calculated using eqn (1).

$$\text{Degradation efficiency(\%)} = \left(1 - \frac{C_t}{C_0}\right) \times 100 \quad (1)$$

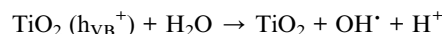
where C_0 is the initial absorbance and C_t is the final absorbance of MB with respect to time.

2.3.1 Reaction mechanism. The structure of semiconductors plays a significant role in photocatalysis. The mechanism of TiO₂ as a photocatalyst is described below and depicted in Fig. 2.

(i) Photoexcitation: photocatalysis (PCA) begins with photoexcitation, which occurs when photons have energy equal to or greater than the band gap energy of the material (E_g). These photons urge electrons in the VB to move towards the CB.

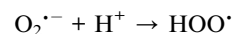
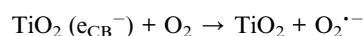


(ii) Water ionization: in order to form a hydroxyl radical (OH[•]), the positive holes (h⁺) can oxidize hydroxide ions or water molecules that have been adsorbed on the surface of TiO₂.



At the photocatalyst surface, these OH[•] radicals act as oxidizing agents that strike adsorbed organic molecules that are very close to the catalyst surface and take part in mineralization to improve decontamination.

(iii) Ionosorption of oxygen: superoxide radicals (O₂^{•-}) are formed when excited electrons react with oxygen. Protons and superoxide radical anions combine to generate hydroperoxyl radicals (HOO[•]).



O₂^{•-} not only plays a part in more oxidation cycles but also prevents e⁻/h⁺ recombination while keeping TiO₂ a neutral molecule.

(iv) Photosensitized oxidation: in conditions where catalysts are present, the excited state of MB can inject an electron into the conduction band. After this step, the dye is changed into a cationic radical, which subsequently degrades and produces compounds in accordance with the following reactions:

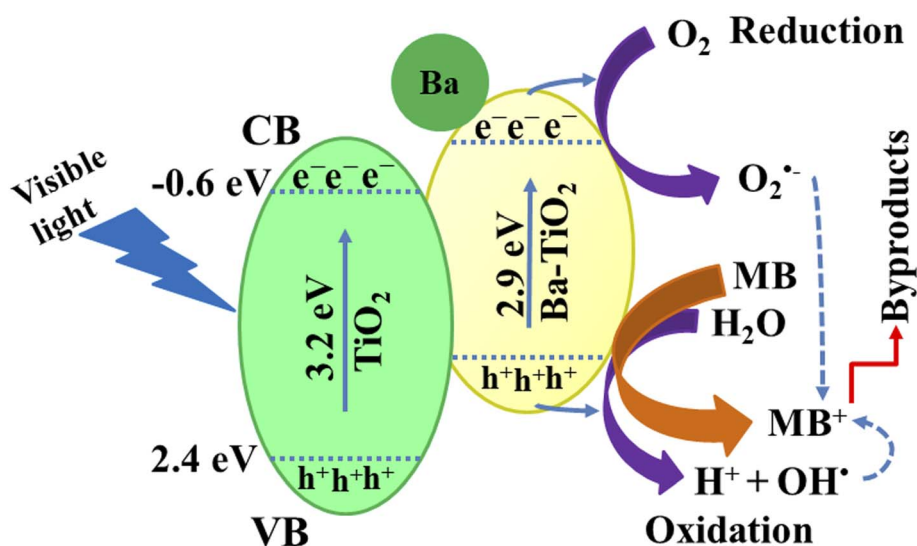
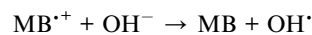
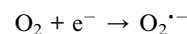
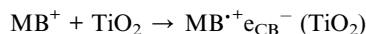
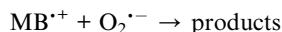


Fig. 2 Schematic photocatalytic mechanism of the Ba-doped TiO₂ QDs.





When the catalyst absorbs visible light holes in the valence band (VB), electrons are produced in the conduction band (CB). In this way, electron-hole pairs are generated and initiate the redox reaction on the TiO_2 surface. OH^\bullet radicals are produced whenever the holes (h_{VB}^+) interact with an adsorbed hydroxyl group. In contrast, the reaction of electrons (e_{CB}^-) with oxygen produces the superoxide anion radical $\text{O}_2^{\bullet-}$. The degradation of MB occurs due to a collaborative effort of OH^\bullet and $\text{O}_2^{\bullet-}$ radicals. Lastly, the oxidation of organic molecules produces CO_2 and H_2O .²⁷

3. Results and discussion

XRD was employed to examine the crystalline structure and crystallite size of undoped and doped TiO_2 in the range of 15° – 65° (Fig. 3a). The identified TiO_2 pattern is consistent with JCPDS card no. 00-021-1272, revealing that the peaks coincide with the anatase phase of TiO_2 . The tetragonal structure was validated by the diffraction peaks detected at $2\theta = 25.28^\circ$, 37.80° , 48.05° , 53.89° , and 62.69° , which synced with the (101), (004), (200), (105), and (204) planes, respectively.²⁸ Additionally, the presence of barium titanium oxide ($\text{Ba}_2\text{O}_{20}\text{TiO}_9$) was observed at 20.0° , 27.4° , 28.6° , and 44.92° , with the hkl values of (013), (141), (220) and (303), respectively, which matched well with JCPDS card no: 00-040-0405. Upon Ba doping, intense TiO_2 peaks were found and peak shifting was observed towards higher angles, which indicates an improvement in crystallinity. The crystalline sizes of the synthesized dopant-free and doped TiO_2 were computed using the Debye-Scherrer formula.

Subsequently, SAED patterns were recorded to explore the highly crystalline nature of the pristine and Ba-doped TiO_2 , as depicted in Fig. 3c–e. The XRD results validate the tetragonal crystalline structure of TiO_2 by indexing the images with the diffraction planes (101), (004), (105), (200), and (204).

FTIR analysis was applied for the functional group identification of the doped TiO_2 samples. Wavenumbers ranging from 4000 to 500 cm^{-1} were used to record the spectra (Fig. 3b). The Ti–O–Ti bridging stretching and Ti–O stretching modes were observed at the bands at 527 and 628 cm^{-1} , respectively.²⁹ Bands appeared at 1640 cm^{-1} and 3200 – 3400 cm^{-1} , corresponding to the hydroxyl group (OH) bending and stretching vibrations caused by the presence of H_2O molecules in the TiO_2 catalyst.³⁰ Band ranges at 2150 – 2360 cm^{-1} were revealed with higher doping concentrations of Ba, ascribed to the physical adsorption of CO_2 and CO .³¹ The spreading of the adsorption region over 950 – 400 cm^{-1} upon Ba doping was attributed to the characteristic M–O bond oscillations (where M = Ti and Ba). Shifting in the band position suggests that Ba^{2+} ions were successfully doped into TiO_2 .³²

The optical properties of TiO_2 are widely known to be highly dependent on the material type (e.g., single-crystal, powder) along with the fabrication methods. The absorbance spectra of undoped and barium-doped TiO_2 were measured by UV-vis spectrophotometry. The absorption peak identified for the as-prepared TiO_2 at $\sim 350\text{ nm}$ is attributed to electron transfer from the 2p to 3d region of the VB of O^{2-} to the CB of Ti^{3+} .³³ In the absorption spectra, a redshift is observed with Ba (2%, 4%) doping, resulting in decreased band gap energy (E_g). Tauc transformation was used to calculate E_g , as displayed in Fig. 4b. The estimated E_g was 3.25 eV for TiO_2 ,³³ and E_g was reduced with a low and high doping concentration of Ba (3.16 eV and

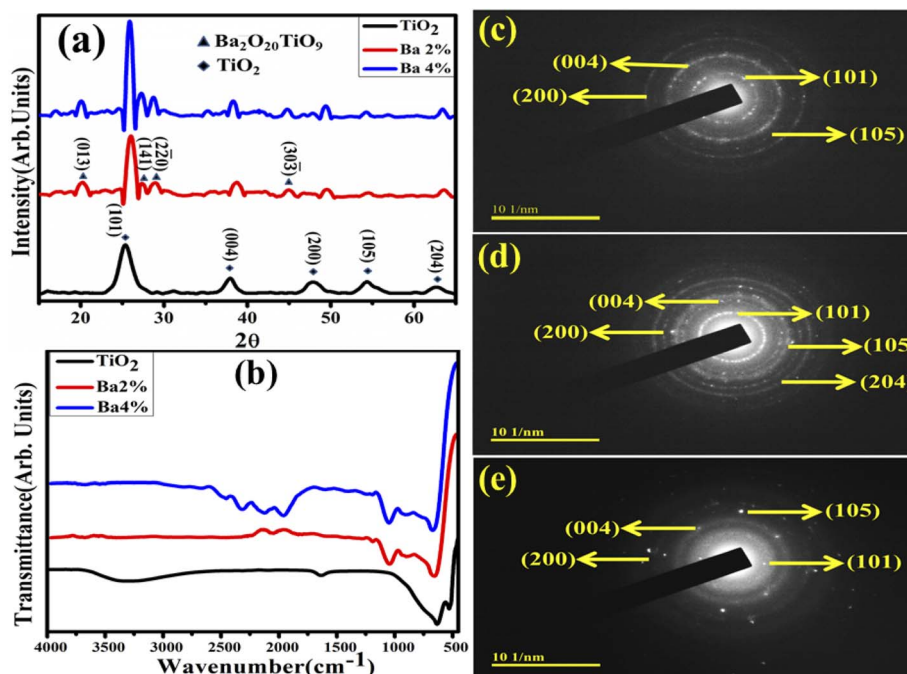


Fig. 3 (a) X-ray diffraction patterns, (b) FTIR spectra and (c–e) SAED patterns of pure and Ba (2%, 4%)-doped TiO_2 QDs.



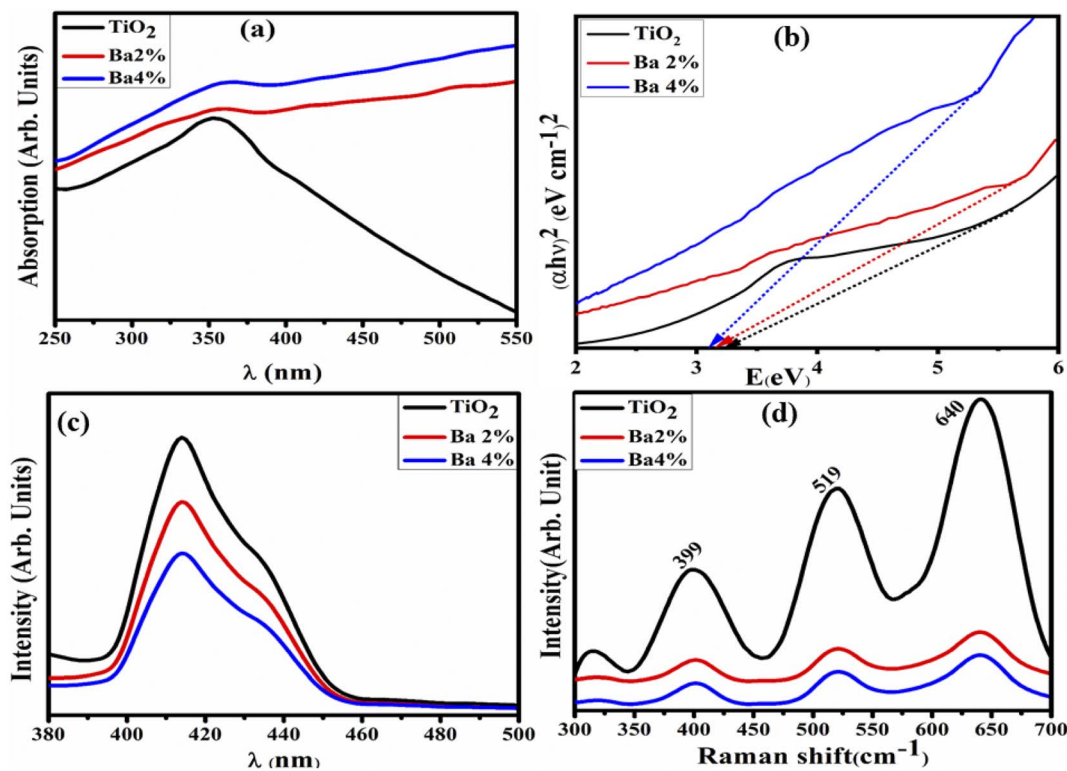


Fig. 4 (a) Absorbance spectra, (b) E_g by Tauc plot, (c) photoluminescence spectra and (d) Raman spectra of the TiO_2 and Ba (2%, 4%)-doped TiO_2 QDs.

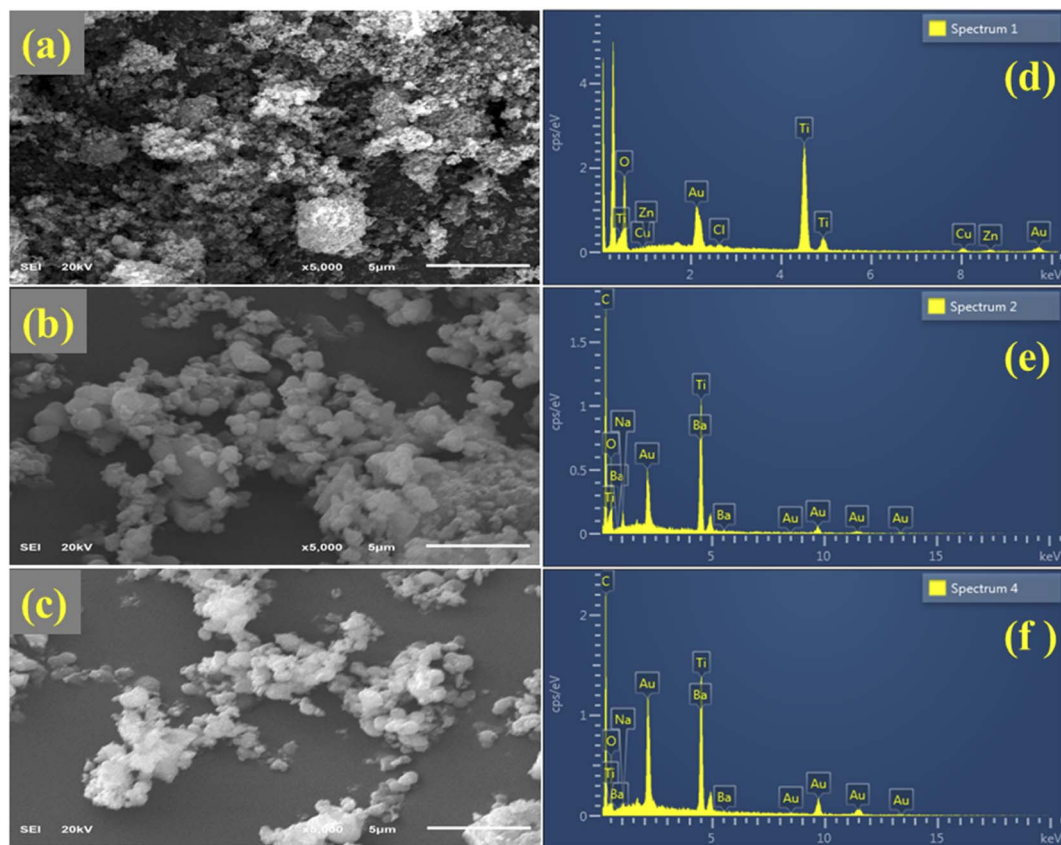


Fig. 5 (a–c) FESEM images and (d–f) EDS analysis of pure and Ba (2%, 4%)-doped TiO_2 .



3.12 eV, respectively). The reduction in the band energy gap can be attributed to the increasing crystallite size, as demonstrated by the XRD measurements.

PL spectroscopy was used to assess the e^-h^+ recombination rate, recorded between 380 and 500 nm at an excitation wavelength of 350 nm (Fig. 4c). The TiO_2 peak intensity observed at 414 nm demonstrated the high recombination rate of e^-h^+ pairs, which influences the TiO_2 photocatalytic performance significantly. Furthermore, the peak intensity was reduced, which indicates a lower charge recombination rate upon doping and thus higher photocatalytic activity.^{34,35} These emission transitions have been credited to unilaterally charged oxygen vacancies in TiO_2 and arise when a photogenerated h^+ mixes with an e^- taking up oxygen vacancies.³⁶ When the concentration of Ba increases, the peak intensity reduces as higher concentrations restrict the inter-nuclear space, and the emission intensity begins to flow towards the energy-killing domains.^{37–40} Raman spectroscopy was used to investigate the lattice strain, phase transformations, lattice dynamics, and local symmetry of the samples.^{41,42}

Fig. 4d presents the Raman spectra of doped TiO_2 between 300 and 700 cm^{-1} . The peaks appearing at 399, 519, and 640 cm^{-1} for the samples are attributed to distinct LO vibration modes, demonstrating the improved crystallinity of TiO_2 with Ba (2%, 4%) doping, having minimum interlayer stress and defect density.⁴³ All the samples exhibited high bands at 399,

519, and 640 cm^{-1} , which most certainly correlate to the permissible Raman modes B_{1g} , B_{1g} , and E_g , respectively, with the space group D_{4h} .^{19,44}

The surface morphology and chemical composition of the synthesized QDs were confirmed by FESEM and EDS spectroscopy, as shown in Fig. 5. Aggregated quantum dots (QDs) of different sizes appear (Fig. 5a). Upon Ba doping, small-scale agglomeration was observed (Fig. 5b). A layer of Ba is formed as the concentration increases, and the QDs are dispersed (Fig. 5c). The EDS spectra for the TiO_2 QDs are presented in Fig. 5d–f. Prominent peaks of Ti with O reveal the successful preparation of TiO_2 QDs. The presence of Ba in the spectra confirms the successful incorporation of dopants, whereas the peaks of Cu and C are attributed to the Cu grid and carbon tape used to mount the samples, respectively (Fig. 5e and f). Furthermore, Au, Zn, and Cl peaks emerged, which were assigned to the coating of gold sprayed over the sample and some contamination. Additionally, a peak of Na was found in each prepared sample, ascribed to the use of NaOH to maintain the pH of the solutions.

The morphology, size, and shape of the TiO_2 and Ba (2%, 4%)-doped TiO_2 QDs were examined using TEM, as illustrated in Fig. 6a–c. Spherical-shaped small-size QDs ranging from 6 to 9 nm are observed, as shown in Fig. 6a. On doping with Ba, the synthesized QDs appeared to be loosely bound with the TiO_2 QDs (Fig. 6b). By increasing the dopant concentration, the QDs

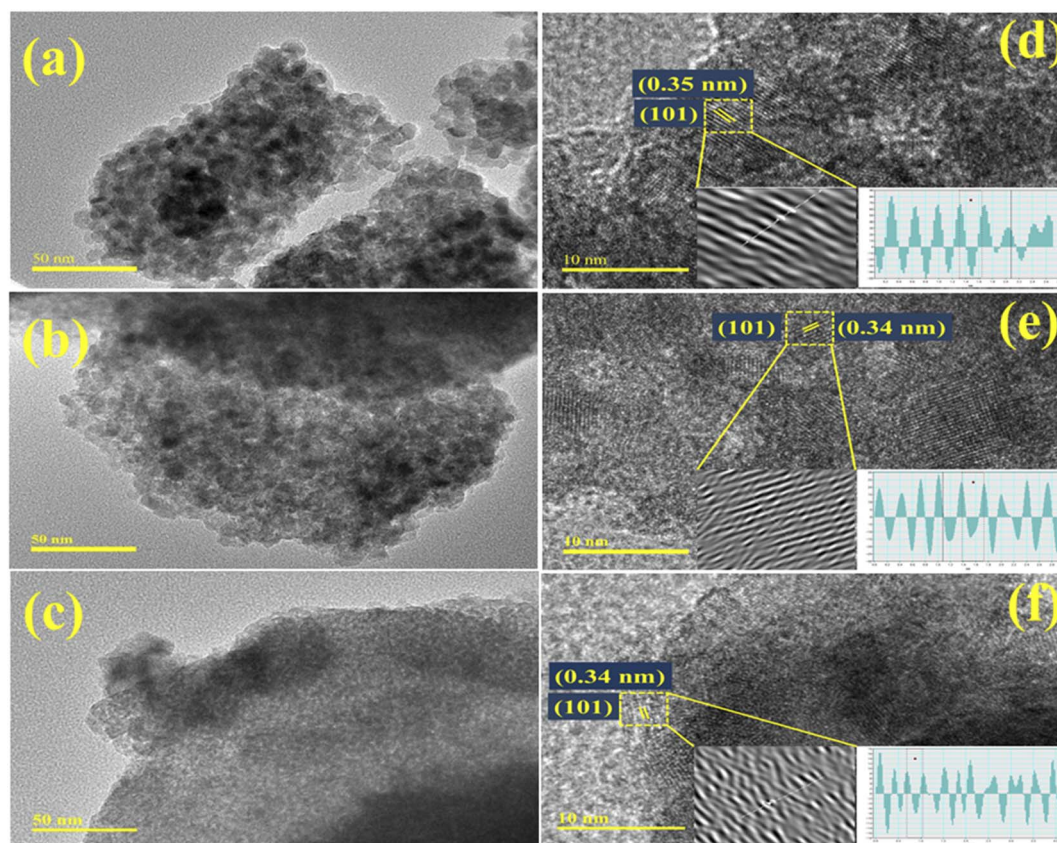


Fig. 6 (a–c) TEM images of pure and Ba (2%, 4%)-doped TiO_2 ; (d–f) interlayer d -spacing HRTEM images of undoped and Ba-doped TiO_2 .



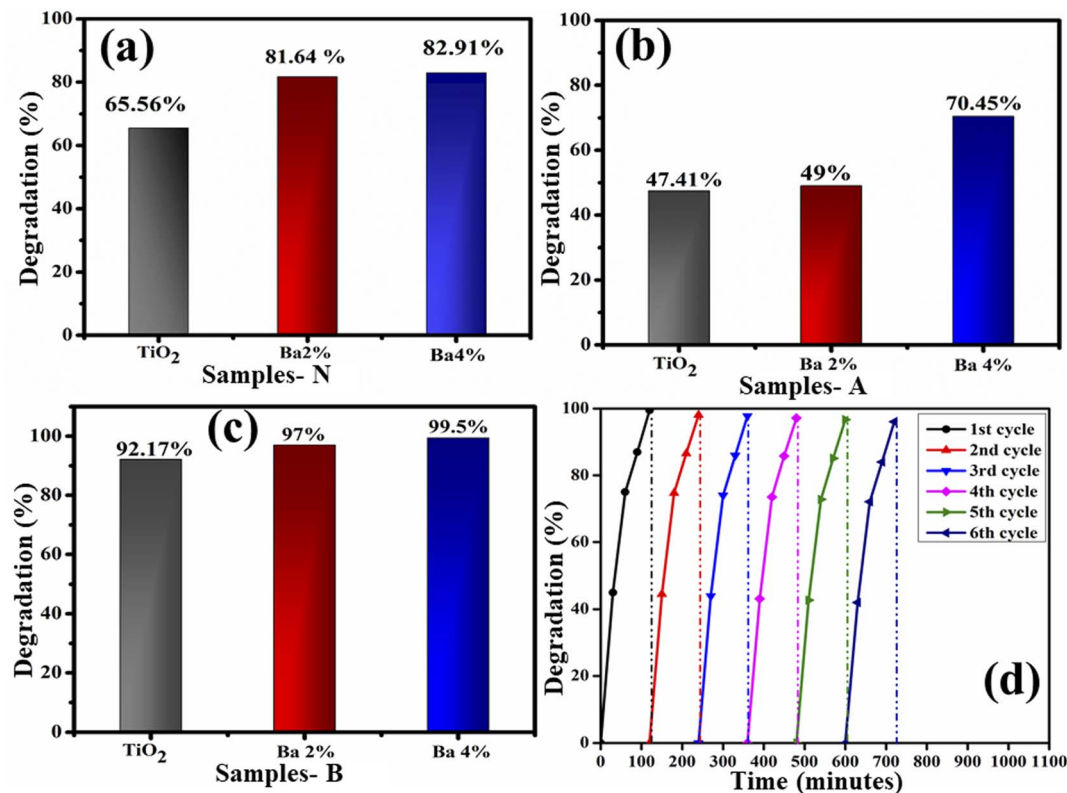


Fig. 7 Photocatalytic degradation of TiO_2 and Ba (2% 4%)-doped TiO_2 in (a) neutral, (b) acidic and (c) basic media and (d) the photocatalytic degradation of MB by 4% Ba-doped TiO_2 .

Table 1 Comparison of the present work with the reported literature

Photocatalysts	Pollutants	Time	Degradation	References
CeO_2 -doped TiO_2	Rhodamine B	8 h	99.89%	50
TiO_2 -TMAOH	MB	20 min	80%	51
TiO_2 thin-films	MB	240 min	92%	52
TiO_2 /PVA	MB	120 min	98.43%	53
TiO_2 /PAAm	MB	5 h	95.00%	54
Ba-doped TiO_2	MB	120 min	99.5%	Present study

seem to become overlapped and highly aggregated, which may result in their decreased size (Fig. 6c). Moreover, HRTEM was utilized to determine the interlayer d -spacing, as shown in Fig. 6d–f. The interlayer d -spacing of TiO_2 and Ba (2%, 4%)-doped TiO_2 were estimated to be 0.35, 0.34, and 0.34 nm, respectively, assigned to the (101) plane for doped and undoped TiO_2 , and well agreed with the XRD results. The above optical, structural, and morphological characterizations for doped TiO_2 indicate that the Ba-doped TiO_2 lattice has an anatase phase with a tetragonal crystalline structure, as confirmed by SAED. The addition of Ba increased the absorption and introduced a redshift as the crystallite size increased, which could capture a wide range of wavelengths. The morphological study confirmed the formation of QDs with agglomeration upon doping.

To determine the PCA of the prepared nanocatalyst, UV-vis spectrophotometry was used (Fig. 7). The degradation

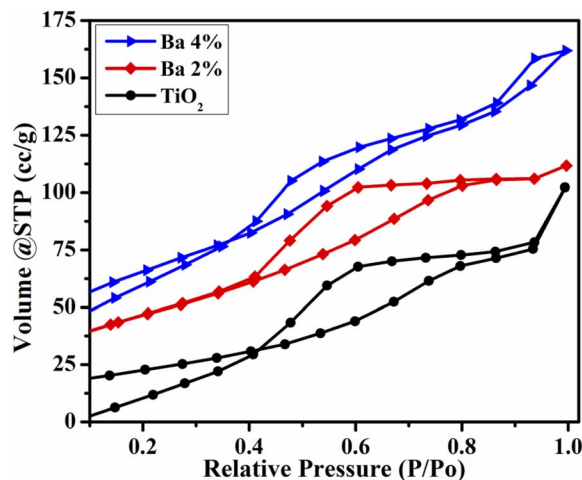


Fig. 8 N_2 adsorption–desorption isotherms.



efficiency was calculated against acidic, basic and neutral media. In a neutral medium (pH 7), the percentage degradation was estimated to be 65.56%, 81.84%, and 82.91% for dopant-free and doped TiO₂, respectively, as presented in Fig. 7a. The solution pH influences the photocatalytic mechanism, which may affect the adsorption of the dye on the photocatalyst surface. MB, as a positively charged dye, degrades slowly at lower pH; as a result, the degradation efficiency is enhanced on increasing the pH. Moreover, in an acidic medium (pH 4), the

measured degradation results are 47.41%, 49%, and 70.41% for the doped and undoped TiO₂, respectively, as demonstrated in Fig. 7b. From the results in the acidic medium, the degradation of MB seems to be ineffective compared to that in the neutral medium, attributed to the positively charged surfaces of the catalysts, which tend to resist the adsorption of the cationic adsorption species. Furthermore, in a basic medium (pH 12), the prepared nanocatalyst exhibits efficient dye degradation results of 92.17%, 97%, and 99.5% for TiO₂ and Ba (2%, 4%)-doped TiO₂, respectively, as illustrated in Fig. 7c. Increased electrostatic interaction between the positively charged dye and the negatively charged catalyst in the basic dye solution causes the surface charges to become negative, which facilitates dye degradation.^{45–47} Generally, the PCA is influenced by the surface area of the nanocatalyst, as a large surface area provides more active sites, increases the number of redox reactions on the nanocatalyst surface and results in the degradation of MB.^{48,49} The results show that the MB dye degrades maximally when treated with Ba-doped TiO₂ rather than with the TiO₂ QDs. For the Ba-doped TiO₂ photocatalysts, the possible charge separation mechanism is described as follows. Charge pairs are formed when visible light interacts with the TiO₂ QDs, and electrons in the charge pairs move from the valence band (VB) to the conduction band (CB), leaving a hole in the VB. Most of these electron–hole pairs tend to recombine and lower the

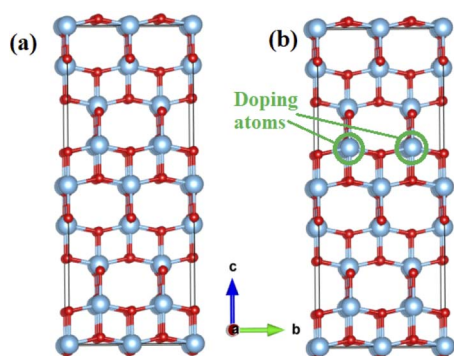


Fig. 9 Computational model (a $2 \times 2 \times 2$ supercell) for the anatase (a) TiO₂ and (b) Ba-doped TiO₂.

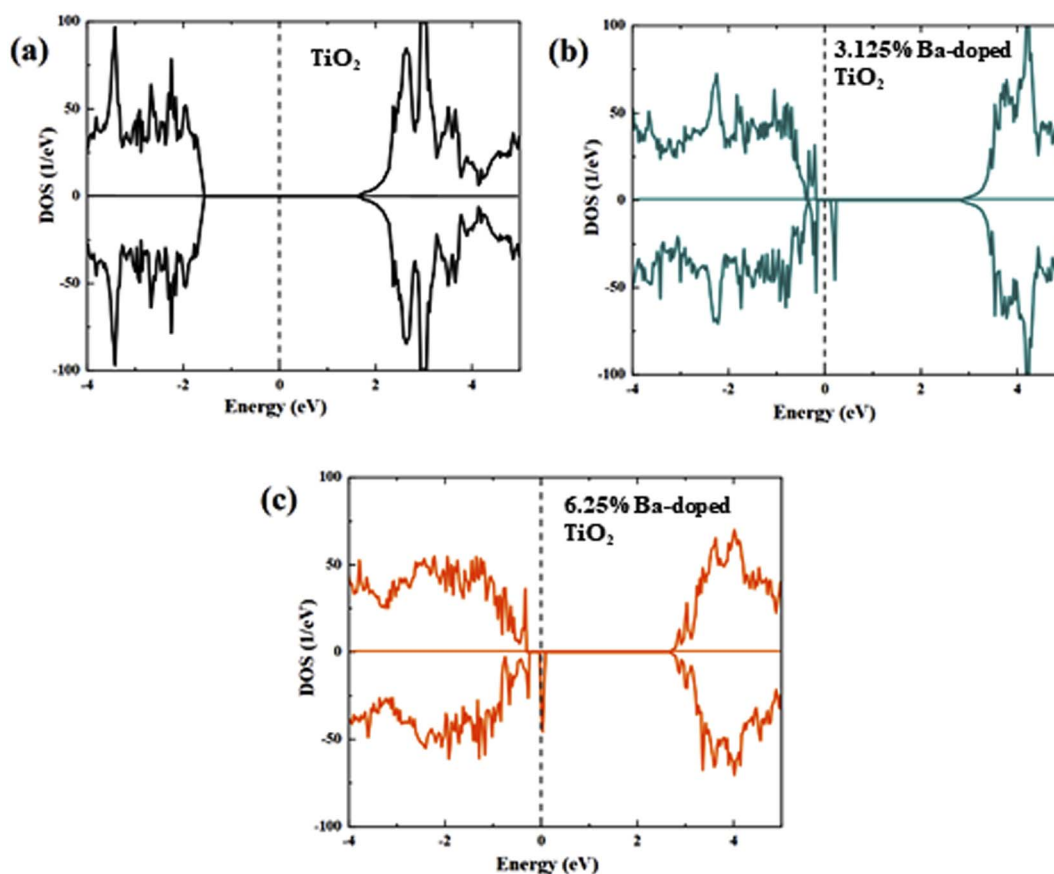


Fig. 10 Calculated total of (a) pure, (b) 3.125% Ba- and (c) 6.25% Ba-doped TiO₂ using the DFT-1/2 method. The Fermi level is represented by the red vertical dashed line.



degradation capability of TiO₂. Incorporating metal ions (Ba) into the TiO₂ lattice as electron traps inhibits the recombination of charge pairs, resulting in increased photocatalytic activity. The separation of charges and the transport of these charges play a significant role in enhancing the photocatalytic activity.²⁵ In this regard, 4% Ba-doped TiO₂ shows the maximum results in all the media for MB degradation, attributed to the lower recombination rate as evidenced from the PL spectra (Fig. 4c). 4% Ba-doped TiO₂ has lower charge recombination that results in enhanced photocatalytic degradation. Additionally, recycling experiments were used to examine the photostability of TiO₂ doped with 4% barium, as presented in Fig. 7d. Even after being subjected to six separate cycles of the photocatalytic degradation of MB by visible light, the high-concentration Ba-doped TiO₂ QDs exhibited a high level of stability and maintained an exceptional level of activity. A literature comparison with the current study is provided in Table 1.

The BET surface area and pore size of the as-prepared TiO₂ and Ba-doped TiO₂ samples were investigated using BET analysis. The N₂ adsorption-desorption isotherms of all the samples are shown in Fig. 8. The adsorption-desorption isotherm of TiO₂ and Ba-doped TiO₂ matched the characteristics of a traditional type IV curve that had an H4 hysteresis loop (Fig. 8). A mesoporous structure was found in both the undoped and doped TiO₂ QDs. In addition, the BET surface areas of the

prepared QDs increased gradually to 119.183, 132.846, and 138.5 m² g⁻¹ for TiO₂ and Ba (2%, 4%)-doped TiO₂, respectively. The average pore size was calculated to be 3.6, 4.5, and 10.8 nm for dopant-free and doped TiO₂, respectively. It can be observed that the incorporation of Ba increased the surface area, which is beneficial for enhanced photocatalytic activity.

3.2 Electronic calculations

The electronic structures and optical characteristics were investigated by comprehensive first-principles calculation with the DFT method using the QuantumATK software package.⁵⁵ In our calculation, the exchange and correlation potential were dealt with by functionals at both the Perdew–Burke–Ernzerhof (PBE)⁵⁶ and DFT-1/2 method levels.^{57,58} The Brillouin region was represented by a Monkhorst–Pack special *k*-point grid (5 × 5 × 1 for geometric optimization and 10 × 10 × 4 for electronic structure calculation). The basis set was set to the norm-conserving PseudoDojo⁵⁹-medium with a mesh cut-off energy of 125 Ha. The force tolerance was set to 0.01 eV Å⁻¹ for geometric optimization.

The crystalline atomic structure of anatase TiO₂ was modeled by a 2 × 2 × 2 supercell containing 32 Ti atoms and 64 O atoms, as illustrated in Fig. 9a. The calculated lattice constants for TiO₂ (*a* = *b* = 3.782 Å and *c* = 9.552 Å) are very close to the experimental data (*a* = *b* = 3.7842 Å, *c* = 9.5146 Å).⁶⁰

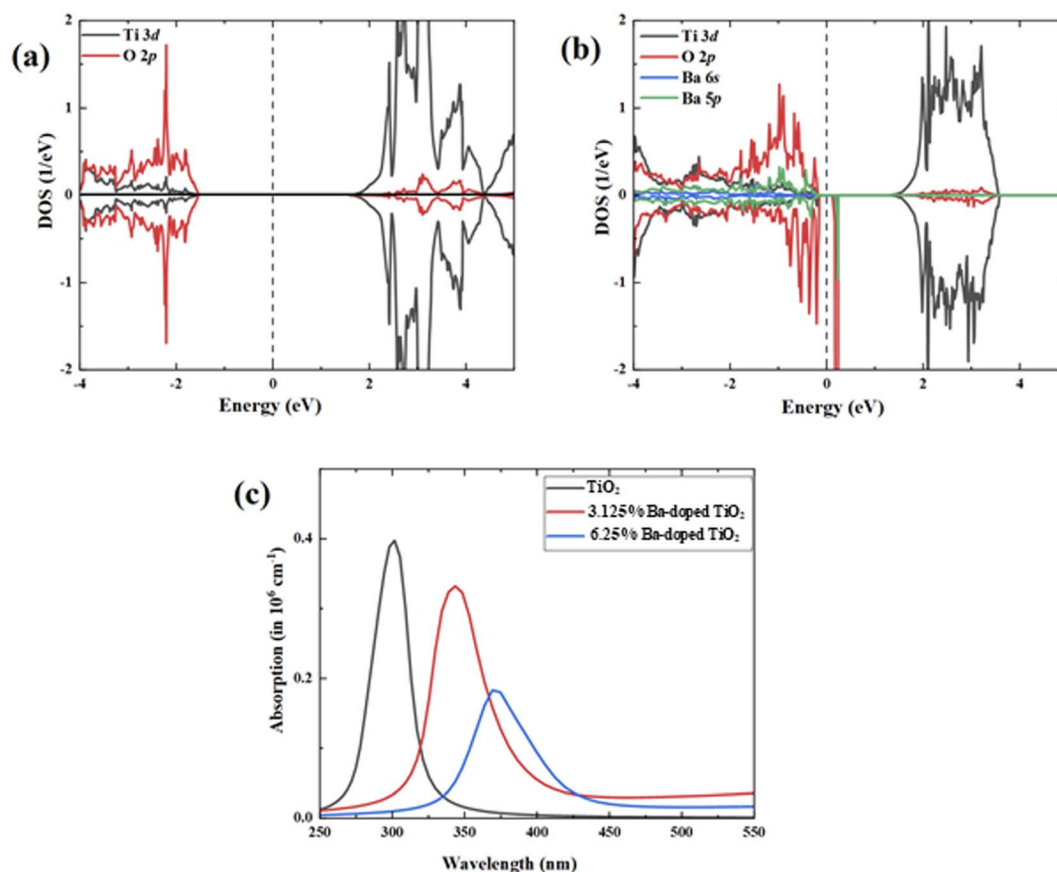


Fig. 11 PDOS of (a) pristine TiO₂ and (b) 3.125% Ba-doped TiO₂ and (c) the optical absorption spectra.



The Ba-doped structures were prepared by substituting one and two Ti atoms with one and two Ba atoms, affording doping concentrations of 3.125% (1/32) and 6.25% (2/32), respectively (Fig. 9b). When Ba impurities were introduced into the TiO₂ structure, the *a*- and *c*-lattice values (*a* = *b* = 3.790 Å, *c* = 9.766 Å and, *a* = *b* = 3.778 Å *c* = 10.013 Å for 3.125% and 6.25% Ba-doped TiO₂, respectively) were slightly higher than those in the pristine material. The hexagonal lattice deformation due to Ba-doping causes this difference.

The electronic densities of states of the undoped and doped systems were calculated to analyze the effect of Ba doping into the TiO₂ lattice from the perspective of the electronic structures (Fig. 10). The calculated pristine TiO₂ band gap was found to be 3.17 eV based on the GGA-1/2 method, which is very close to our measured experimental value and previously reported results.^{61–65} The analysis of the partial DOS of pure TiO₂ (Fig. 11a) shows that the upper valence region is mostly dominated by O 2p states, while the lower conduction region is mainly due to the Ti 3d character. Doping Ba at Ti sites modified the electronic structure by generating a slightly asymmetric DOS near the valence band, as illustrated in Fig. 11b. It is observed that in-gap states appear in the band gap, especially in the peak in the DOS of the minority spin close to the upper part of the valence band. It can also be seen that the edge of the valence band advances closer to the Fermi level, indicating an increase in charge-carrier concentration and a modest reduction in the band gap.

The obtained band gap values are found to be 3.07 eV and 3.002 eV for 3.125% Ba- and 6.25% Ba-doped TiO₂, respectively. Additionally, the band gap values are shown to slightly decrease with further increase in Ba concentration. The same behavior was observed in our experimental measurements.

Moreover, the states that appear in the band gap can help to stimulate valence electrons to the conduction band, boosting the visible light absorption, which is beneficial to improving the optical absorption efficiency. The in-gap states may lower the threshold for photoexcitation, decrease the energy needed for electron transitions, and thereby broaden the optical absorption spectrum without altering the energy of the electrons or holes. This result gives a good explanation of the redshift. The features of the PDOS of Ba-doped TiO₂ systems show that the Ba 5p states are coupled with O 2p, while partially appearing in the band gap. The absorption spectra of the pristine and Ba-doped TiO₂ were computed and are illustrated in Fig. 11c. The absorption spectrum analysis displays that the pristine TiO₂ only absorbed photons in the ultraviolet range with no photo-absorption activity in visible light. In the case of the Ba-doped TiO₂ configurations, there is a considerable redshift of the absorption edge owing to the presence of an in-gap near the valence band, which makes it simpler for electrons to be driven to the conduction band.

4. Conclusion

The co-precipitation technique was employed to synthesize TiO₂ and Ba (2%, 4%)-doped TiO₂ quantum dots. The XRD patterns revealed a tetragonal structure and anatase phase with a highly

crystalline nature of TiO₂, as confirmed by SAED. The FTIR spectra present a band at 670 cm⁻¹ attributed to the Ti–O–Ti bond after preparing Ba-doped TiO₂. The absorbance increased with an increasing amount of Ba accompanied by a redshift and decrease in the band gap energy. PL measurement showed a higher electron recombination rate as the intensity increased upon doping relative to the doped TiO₂. The strongest Raman signal of TiO₂ at 640 cm⁻¹ was indexed to the *E_g* vibrational mode utilizing Raman spectroscopy. Moreover, TEM images confirmed the QD formation and the *d*-spacing (0.34–0.35 nm) was evaluated using the Gatan software, comparable with the XRD measurements. Additionally, the dye degradation efficiency of the prepared nanocatalysts was examined against MB using different media. Among these, the most effective dye removal efficiency was observed in the basic medium. To investigate the influence of Ba doping on the TiO₂ lattice from the viewpoint of the electronic structures, the electronic densities of states of the undoped and doped TiO₂ systems were computed. With increasing the Ba concentration, the band gap values appear to slightly decrease. According to the absorption spectrum study, pristine TiO₂ solely absorbed photons in the ultraviolet region, with no photo-absorption activity in visible light. The existence of an in-gap in the valence band in the Ba-doped TiO₂ configurations causes a significant redshift of the absorption edge, making it easier for electrons to be pushed to the conduction band. Lastly, the TiO₂ and Ba (2%, 4%)-doped TiO₂ QDs were found to be more effective against MB dye removal (99.5% in basic medium).

Conflicts of interest

There are no conflicts to declare.

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